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Fermi-Level Engineering of Nitrogen Core-Doped Armchair Graphene Nanoribbons

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ABSTRACT: Substitutional heteroatom doping of bottom-up engineered 1D graphene nanoribbons (GNRs) is a versatile tool for realizing low-dimensional functional materials for nanoelectronics and sensing. Previous efforts have largely relied on replacing C−H groups lining the edges of GNRs with trigonal planar N atoms. This type of atomically precise doping, however, only results in a modest realignment of the valence band (VB) and conduction band (CB) energies. Here, we report the design, bottom-up synthesis, and spectroscopic characterization of nitrogen core-doped 5-atom-wide armchair GNRs (N_2-5-AGNRs) that yield much greater energy-level shifting of the GNR electronic structure. Here, the substitution of C atoms with N atoms along the backbone of the GNR introduces a single surplus π-electron per dopant that populates the electronic states associated with previously unoccupied bands. First-principles DFT-LDA calculations confirm that a sizable shift in Fermi energy (~1.0 eV) is accompanied by a broad reconfiguration of the band structure, including the opening of a new band gap and the transition from a direct to an indirect semiconducting band gap. Scanning tunneling spectroscopy (STS) lift-off charge transport experiments corroborate the theoretical results and reveal the relationship among substitutional heteroatom doping, Fermi-level shifting, electronic band structure, and topological engineering for this new N-doped GNR.
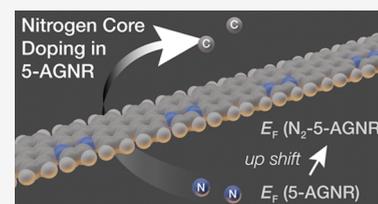
INTRODUCTION

Graphene nanoribbons (GNRs) represent an emerging class of bottom-up synthesized designer quantum materials whose intrinsic electronic structure can be tuned with atomic precision.[1,2] GNRs provide access to new physical properties in molecular nanostructures, e.g., antiferromagnetic ordering,[3] the emergence of symmetry-protected topological phases,[4−7] and band gap engineering.[8−10] Switching between a direct and an indirect band gap, however (e.g., by switching the crystal momentum difference between the valence band (VB) maximum and conduction band (CB) minimum), has not been accomplished.[11,12] Such control over relative band alignments in *k*-space would have a dramatic impact on the selection rules for optical transitions, exciton lifetimes, and electron−phonon coupling that are relevant for applications in light-harvesting devices,[13,14] light-emitting diodes,[15−17] and optoelectronic materials.[18,19]

N-doping GNRs along their edges have been shown to typically result in a downward shift in band energies relative to undoped GNRs.[10,20−30] Recently, we reported a general approach for embedding a periodic lattice of trigonal planar N atoms into the backbone of bottom-up-synthesized chevron-type GNRs (cGNRs).[31] The substitution of trigonal planar C atoms by N atoms introduced a single surplus π-electron per N atom that populated emergent dopant states within the semiconducting gap. In this case, charge transfer between the highly localized dopant states in the cGNR and the underlying Au substrate gave rise to a weakly correlated 1D Kondo chain.

Here, we describe the design and on-surface synthesis of substitutionally nitrogen-doped 5-AGNRs (N_2-5-AGNRs) derived from a new molecular building block. Figure 1A shows a model of the structure of the new extended N_2-5-AGNR. The unit cell comprises a quinoxaline core flanked on either side by naphthalene units fused along the short zigzag segments. This unusual orientation places all N atoms along the same armchair edge of the ribbon, effectively breaking mirror symmetry along the main axis of the 5-AGNR. The implications of this new doping pattern on the electronic structure of 5-AGNRs can be seen in *ab initio* density functional theory (DFT) calculations of the band structure. The band structure of an all-carbon 5-AGNR compared to that of a N_2-5-AGNR is shown in Figure 1B,C. Gray lines at $E − E_F$ = 0.0 and $E − E_F$ = −1.0 eV correspond to the energy of topological end-states (ES) associated with the short zigzag ends of finite 5-AGNR segments for the undoped (0.0 eV) and doped (−1.0 eV) cases.[32] The width of the red lines in Figure 1C represents the relative contribution of orbitals having nitrogen character to the corresponding bands.
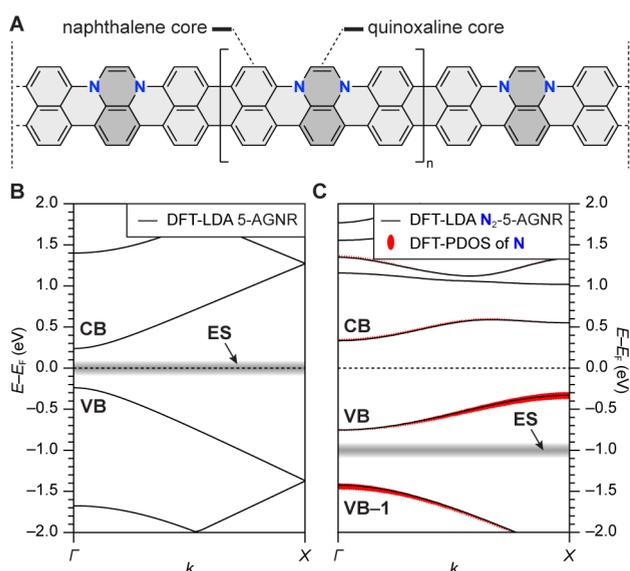
Figure 1. (A) Structure of a N_2-5-AGNR. (B) DFT-LDA-calculated band structure of all-carbon 5-AGNR. (C) DFT-LDA-calculated band structure of a N_2-5-AGNR. The relative contribution from orbitals with nitrogen character is shown by the width of the red lines. Features associated with the topological end-state (ES) (obtained from a calculation on a GNR with five repeating units) are indicated by gray lines.

A direct comparison of the Fermi levels of the 5-AGNR band structure versus those of the N_2-5-AGNR band structure reveals the effect of adding two additional electrons per unit cell due to N atom doping. N_2-5-AGNR states are seen to be occupied, which can be associated with the highly dispersive CB of the all-carbon 5-AGNR. This band now comprises the new N_2-5-AGNR VB and has a reduced width of ~0.4 eV. The new VB is dominated by contributions from orbitals having carbon character at $k = \Gamma$ (bottom of the VB) and nitrogen character at $k = X$ (top of the VB). The previous degeneracy at $k = X$ is now lifted, and the k-location of the valence band maximum is reversed compared to the 5-AGNR. The E_F of N_2-5-AGNRs lies between the top of the VB at $k = X$ and the bottom of the CB at $k = \Gamma$, thus producing an indirect gap of ~0.6 eV. The vertical shift in E_F (~1.0 eV) compared to the 5-AGNR is further reflected in the realignment of the topological ES to $E - E_F = -1.0$ eV.

Besides inducing a direct-to-indirect band gap transformation, the introduction of substitutional N atom dopants brings about a change in the band topology compared to 5-AGNRs. The topological invariant of a GNR with spatial and time reversal symmetry is given by $\mathbb{Z}_2 = \Pi_n M_n(\Gamma)M_n(X)$, where $M_n(\Gamma)$ and $M_n(X)$ are the mirror symmetry eigenvalues for each filled band n at the Γ and X points, respectively.[4,33] If one were to evaluate this product over filled bands to the top of VB−1 (VB) in N_2-5-AGNRs (5-AGNRs), then the $\mathbb{Z}_2$ invariant corresponds to a topologically nontrivial nanoribbon ($\mathbb{Z}_2 = 1$). Because of the new VB for N_2-5-AGNR, however, the topological invariant of this ribbon becomes $\mathbb{Z}_2 = 0$. Shifting E_F via the introduction of nitrogen core dopants in 5-AGNRs is thus also accompanied by a change in the GNR topological order.

RESULTS AND DISCUSSION

Bottom-up Synthesis of N_2-5-AGNRs. The synthesis of the molecular precursor for N_2-5-AGNRs, **1b**, is depicted in Figure 2. Reduction of 4,7-diiodobenzo[c][1,2,5]thiadiazole
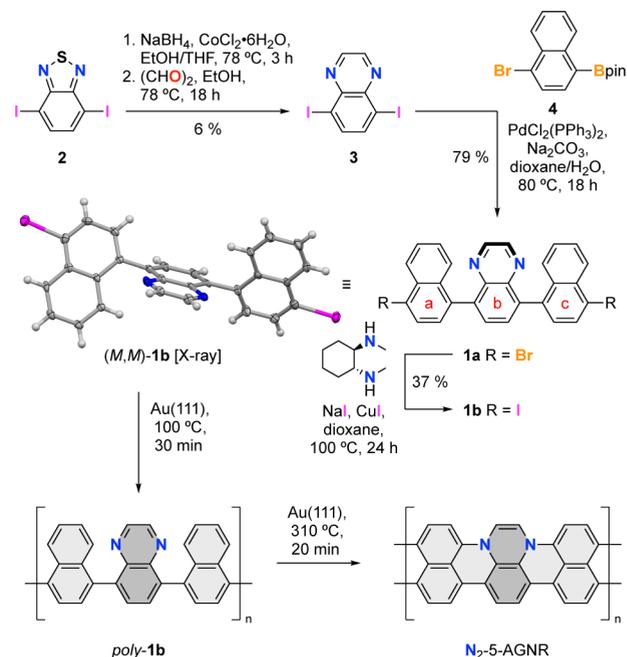

Figure 2. Bottom-up synthesis of N_2-5-AGNR from molecular precursor **1b**. Single X-ray crystal structure of **1b** is shown. Only the (*M,M*) atropisomer is depicted here. Thermal ellipsoids are drawn at the 50% probability level. Color coding: C (gray), I (purple), and N (blue). Hydrogen atoms are placed at calculated positions.

(**2**)[34] followed by condensation with glyoxal yielded 5,8-diiodoquinoxaline (**3**). *Suzuki−Miyaura* cross-coupling of **3** with 2 equiv of 2-(4-bromonaphthalen-1-yl)-4,4,5,5-tetramethyl-1,3,2-dioxaborolane (**4**) yielded the parent dinaphthyl quinoxaline **1a**. A subsequent aromatic *Finkelstein* reaction yielded 11,16-diiodo-1,8-diphenyltetrabenzo[a,c,h,j]phenazine (**1b**) as the desired monomer precursor. Single crystals suitable for X-ray diffraction and surface-assisted growth were obtained by the slow diffusion of MeCN into a saturated solution of **1b** in CHCl₃. In the crystal, **1b** adopts quasi-C_2 symmetric axially chiral geometry. The dihedral angles between rings a−b and b−c range from 63.6 to 70.5°. The unit cell contains both atropoenantiomers (*P,P*)-**1b** and (*M,M*)-**1b**.

The on-surface synthesis of N_2-5-AGNRs from molecular precursor relies heavily on the subtle differences in the activation barrier for brominated and iodinated precursors, **1a** and **1b**, respectively.[35,36] The lower bond dissociation energy (BDE) of C−I bonds ensures that the step growth polymerization proceeds at a low temperature and is unaffected by trace hydrogen atoms generated during the cyclodehydrogenation step that could lead to undesired chain termination. N_2-5-AGNRs were prepared following established surface-assisted GNR growth protocols. Molecular precursor **1b** was sublimed in ultrahigh vacuum (UHV) from a Knudsen cell evaporator onto a Au(111) surface held at $T = 25$ °C. The inset in Figure 3A shows a topographic STM image of a self-assembled island of **1b**. **1b** adopts a quasi-C_s symmetric conformation on the surface reminiscent of a crescent. A bright feature associated with the central quinoxaline ring protruding from the surface is
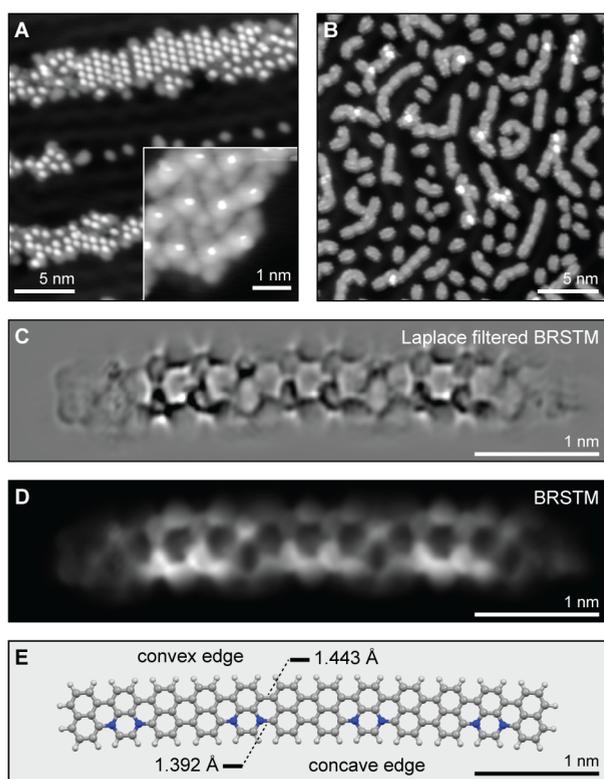
Figure 3. Bottom-up synthesis of N$_2$-5-AGNRs. (A) STM topographic image of polymer intermediates *poly*-**1b** (V_s = −0.1 V and I_t = 80 pA). The inset shows an STM topographic image of a self-assembled island of molecular precursor **1b** on Au(111) (V_s = −0.1 V, I_t = 150 pA). (B) STM topographic images of a sample of N$_2$-5-AGNRs after annealing to 310 °C (V_s = −1 V and I_t = 50 pA). (C) Laplace-filtered bond-resolved STM (BRSTM) image and (D) BRSTM image of a N$_2$-5-AGNR with four repeating units (V_s = 0 V, V_{ac} = 20 mV, and f = 533 Hz). (E) DFT-LDA relaxed structure of a finite N$_2$-5-AGNR tetramer.

flanked on either side by iodonaphthalene units adsorbed nearly coplanar to the underlying substrate.

Annealing the molecule-decorated surface at T_1 = 100 °C for 30 min induces a radical step-growth polymerization that gives rise to polymer intermediate *poly*-**1b** (Figure 3A). A second annealing step at T_2 = 310 °C for 20 min induces a cyclodehydrogenation reaction that leads to the formation of two C−N bonds and one C−C bond per unit cell. Topographic STM images recorded on a low coverage sample (Figures 3B and S1A−C) are dominated by small rectangular structures consistent with the dimension of fully fused monomers (1.2 × 0.68 × 0.18 nm). These smaller molecular structures are interspersed by linear oligomeric nanoribbons featuring a characteristic pattern of periodic constrictions and ranging in length from the dimer (∼2 nm) to the hexamer (∼6 nm). Bond-resolved STM (BRSTM) with a CO-functionalized tip recorded on a tetramer reveals the characteristic internal bonding associated with 5-AGNRs (Figure 3C). An alternating pattern of dark and bright low-bias features is seen along the backbone of the ribbon (Figure 3D). Regions of brighter contrast correspond to 5-AGNR segments resulting from the fusion of two naphthalene units (the perylene core), while the darker features coincide with the position of the N atoms (the quinoxaline core). The subtle curvature of N$_2$-5-AGNRs suggests a preference for the alignment of N atoms along

one armchair edge (the bottom edge of the ribbon depicted in Figure 3C,D). Figure 3E shows the calculated C−N and C−C bond lengths for a N$_2$-5-AGNR tetramer of 1.392 and 1.443 Å, respectively. The periodic bond distortion along the main axis of the ribbon gives rise to the distinctive structure of coexisting convex (all-carbon) and concave (N atom doped) armchair edges. Bond length alternations have previously been observed in other pyrazine-embedded nanographene structures.[31,37]

Electronic Structure Characterization of N$_2$-5-AGNRs. We characterized the electronic structure of N$_2$-5-AGNRs using STM spectroscopy. Figure 4A shows typical dI/dV spectra of a N$_2$-5-AGNR recorded at the positions marked in the inset. Four prominent spectral features appear in the range of −1.5 V < V_s < + 1.5 V. Two peaks centered at V_s = +1.10 V (*Peak* 1) and V_s = +0.06 V (*Peak* 2) are most prominent, along with a shoulder at V_s = −0.4 V (*Peak* 3). The intensities of *Peak* 1 and 2 are largest when the STM tip is placed above the position of a quinoxaline N atom (blue line in Figure 4A). *Peak* 3 is most prominent in spectra recorded above the position of the fused naphthalene units (red line in Figure 4A) and the positions of N atoms. The last spectral feature, a shoulder at V_s = −0.75 V (*Peak* 4), is most prominent near the zigzag end of the N$_2$-5-AGNRs (black line in Figure 4A).

Differential conductance maps recorded at or near the energies of *Peaks* 1−4 (Figure 4B−E) provide further insight into the spatial distribution of the wave function associated with the corresponding electronic states. The dI/dV map at V_s = +1.10 V (*Peak* 1, Figure 4B) shows a distinctive nodal pattern of bright spots emanating from the convex armchair edge that decays across the width of the ribbon and vanishes toward the zigzag ends. A similar characteristic pattern can be observed in a dI/dV map acquired at V_s = −1.90 V (Figures 4F and S2). The latter map exhibits a pair of bright lobes at either end of the ribbon. Differential conductance maps at a bias associated with *Peak* 2 (V_s = +0.06 V; Figure 4C) reveal a nodal pattern that clearly outlines the crystallographic unit cells of a N$_2$-5-AGNR pentamer. Each cell is bounded by a U-shaped bright feature along the concave armchair edge (commensurate with the position of the quinoxaline N atoms) matched on the convex side by two fainter lobes. As the mapping bias is swept across E_F, the nodal pattern evolves into four prominent lobes along the concave edge at V_s = −0.20 V (*Peak* 3, Figure 4D). These are associated with the fused naphthalene units that are mirrored, albeit at a lower signal intensity, across the width of the ribbon. Finally, dI/dV maps recorded at V_s = −0.80 V (*Peak* 4, Figure 4E) show the characteristic signature of a localized N$_2$-5-AGNR end-state.

First-Principles Calculation of N$_2$-5-AGNR LDOS. *Ab initio* DFT calculations enable the rationalization of the spectral features revealed in STS experiments. Figure 4G−K shows the calculated LDOS maps evaluated at the N$_2$-5-AGNR CB minimum, the top and bottom VB edges, the end-states, and the VB−1 edge, respectively. The similarity between theoretically predicted nodal patterns associated with the CB minimum (Figure 4G) and the experimental dI/dV map recorded at V_s = +1.10 V is striking and suggests an assignment of *Peak* 1 to the CB edge. The evolution of the calculated LDOS sampled at the VB top at k = X (Figure 4H) and the VB bottom at k = Γ (Figure 4I) is also mirrored in the experimental dI/dV maps recorded at V_s = +0.06 V (*Peak* 2, Figure 4C) and V_s = −0.2 V (*Peak* 3, Figure 4D), respectively. The gradual increase of density from orbitals with N atom character as k is swept from Γ to X in the theoretical LDOS is
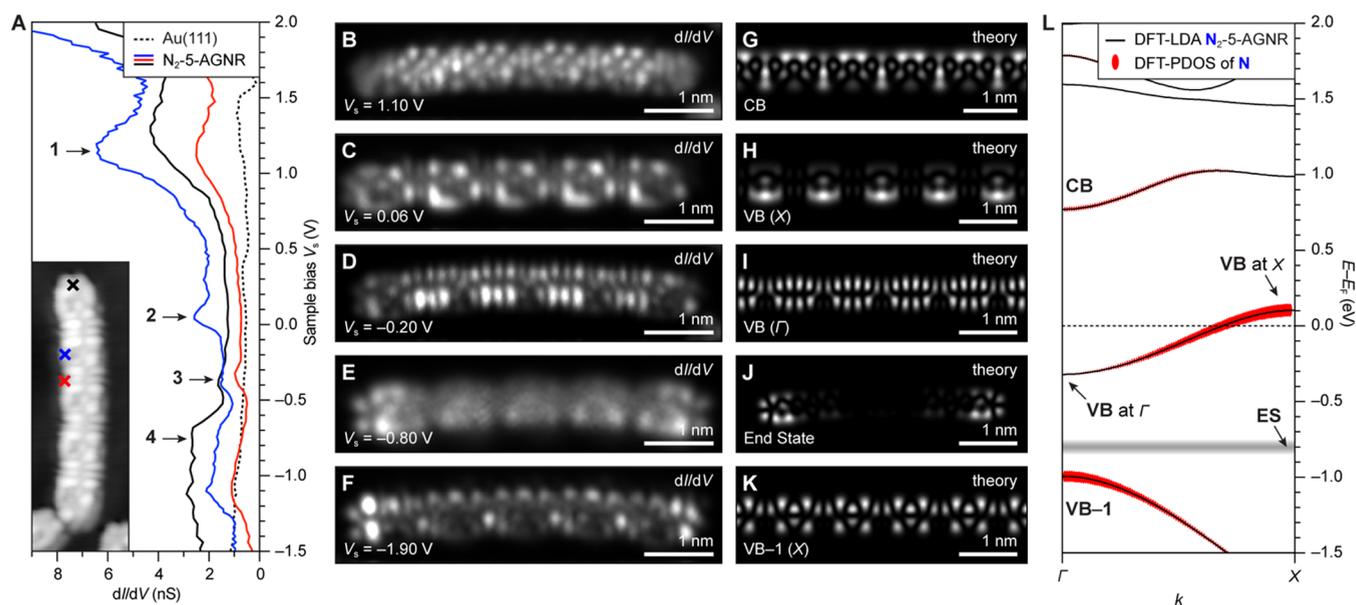
Figure 4. Electronic structure of N_2-5-AGNRs. (A) STS dI/dV spectra recorded on a N_2-5-AGNR with five repeating units at the marked positions in the inset STM topographic image. Prominent features in the dI/dV spectra are labeled as *Peak* 1−4 (spectroscopy: V_{ac} = 4 mV, f = 533 Hz; imaging: V_s = −0.1 V, I_t = 100 pA, CO-functionalized tip). (B−F) Constant height dI/dV maps recorded at the indicated biases (V_{ac} = 20 mV, f = 533 Hz, and a CO-functionalized tip). (G−I, K) DFT-simulated LDOS maps of the CB bottom, VB top, VB bottom, and VB−1 bottom of N_2-5-AGNR. The maps are calculated at the band edge energies. (J) DFT-simulated LDOS map of the end-states of N_2-5-AGNR with five repeating units. (L) DFT-LDA-calculated band structure of N_2-5-AGNR. Features associated with the CB, VB, and ES are indicated by arrows. E_F location is inferred through comparison of spectroscopic GNR features and Au(111) E_F location (see text).
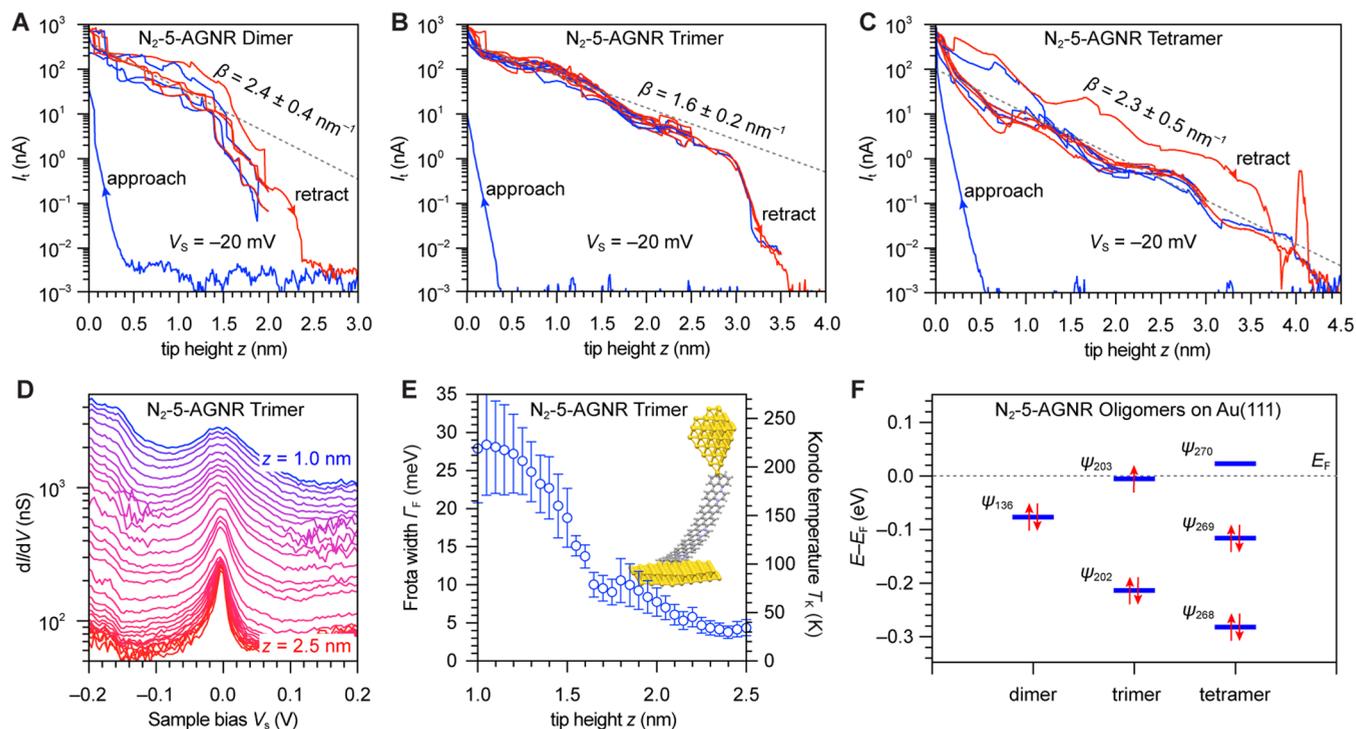

Figure 5. N_2-5-AGNR lift-off experiments. Plot of the tunneling current (I_t) vs. relative tip height (z) for the lifting of a N_2-5-AGNR (A) dimer, (B) trimer, and (C) tetramer at V_s = −20 mV. Color coding: retraction of the STM tip (red), approach of the STM tip (blue). The dashed gray line represents a least-squares fit of the data to exponential decay function $I(z) = I_0 e^{-\beta z}$. (D) STS lift-off experiments performed on a N_2-5-AGNR trimer. Color gradient indicates the apparent z-height of the STM tip above the surface. (E) Frota fit width and Kondo temperature vs. tip height. (F) DFT-calculated molecular orbital energy-level diagram of N_2-5-AGNR dimer, trimer, and tetramer. The dashed lines indicate the position of the E_F of a N_2-5-AGNR on Au(111).

reflected in the increased intensity in dI/dV maps around N atoms as the bias is shifted from V_s = +0.06 V (Figure 4C) to

V_s = −0.20 V (Figure 4D). Similarly, the theoretical LDOS maps associated with the end-states of a finite N_2-5-AGNR

(Figure 4J) and the VB−1 edge at $k = X$ (Figure 4K) are in good agreement with the experimental maps recorded at $V_s = -0.80$ V (Figure 4E) and $V_s = -1.90$ V (Figure 4F).

The picture that emerges from a comparison between theory and experiment for the electronic structure of N$_2$-5-AGNRs on Au(111) is shown in Figure 4L. To match the energy levels of valence band onsets between experimental DOS in Figure 4A and theoretical DOS in Figure 4L, the DFT-calculated band structure of Figure 1C has been rigidly shifted up in energy so that E_F lies 0.1 eV below the VB maximum to model hole-doping from the Au(111) substrate (see Supporting Information Figure S3 for calculated DOS). While DFT-LDA calculations are known to underestimate the size of the band gap, the gold substrate itself introduces a screening of the GNR that partially compensates for the underestimation of the band gap by DFT-LDA. The characteristic topological end-state of a finite, all-carbon 5-AGNR that would normally lie at E_F is now found 0.8 eV below E_F (i.e., *Peak* 4) due to the additional electron density arising from the N-dopants. This is consistent with the close match between the experimental dI/dV maps of Figure 4B−F and the theoretical LDOS distributions shown in Figure 4G−K. The lack of any observable end-states in the energy range between the VB maximum and CB minimum is also consistent with the change in topological character in N$_2$-5-AGNRs from nontrivial (in all-carbon 5-AGNRs) to trivial.

Lift-Off and Charge Transport Experiments. To gain insight into how the electronic structure of a N$_2$-5-AGNR affects its electronic transport behavior, we performed lift-off through-conductance experiments. Here, one end of a N$_2$-5-AGNR is attached to the STM tip and gradually lifted from the surface while recording the change in tunneling current (I_t) at a constant sample bias ($V_s = -20$ mV).[38−40] Figure 5A−C shows the evolution of I_t as a function of tip height (z) for a N$_2$-5-AGNR dimer, trimer, and tetramer, respectively. Each trace shows that within a narrow window defined by the length of the ribbon, I_t decays exponentially with lifting height. The large hysteresis seen for all three lengths indicates that the end of the tip is no longer in contact with the GNR. Since the energy levels of oligomeric N$_2$-5-AGNRs are discrete, a resonant conduction path at this small bias voltage is unlikely.[38,41] Transport through these short ribbons is therefore described as off-resonant tunneling with a characteristic current decay described by $I(z) = I_0 e^{-\beta z}$, where $z = 0$ nm is the reference height, where $I_0 = 700$ nA. A fit of the exponential decay function $I(z)$ to the experimental data in Figure 5A−C (dashed gray lines) reveals characteristic tunneling decay constants (β) for the dimer (2.4 ± 0.4 nm^{-1}), trimer (1.6 ± 0.2 nm^{-1}), and tetramer (2.3 ± 0.5 nm^{-1}) (see Supporting Information, Figure S4, for monomer data). The β-values for the dimer and tetramer are similar in magnitude, but the decay constant for trimers was consistently smaller and warranted further investigation.

Figure 5D shows a series of dI/dV(V) spectra recorded on a N$_2$-5-AGNR trimer. A broad spectral feature centered at $V_s = 0.0$ V is seen to evolve into a single narrow peak as the tip is retracted from $z = 1.0$ to 2.5 nm. The z-independent position of the peak at zero bias and the narrowing of the peak width with lifting height provide evidence that this feature is a Kondo resonance.[32,42−44] Here, a magnetic moment (e.g., an unpaired electron) hosted by the GNR is screened by itinerant electrons in the Au(111) surface, giving rise to zero-energy spin-flip excitations reflected in the increased dI/dV signal at $V_s = 0.0$

V.[45,46] A conventional resonance peak would likely shift away from zero during the highly perturbative lift-off procedure. No analogous features were observed in dI/dV spectra recorded on N$_2$-5-AGNR dimers and tetramers (Supporting Information, Figure S5).[32,39] The curves in Figure 5D were fitted by the Frota function:[47,48]

$$T_F(\varepsilon) = T_0 - A \cdot Re\left[e^{iq} \sqrt{\frac{i\Gamma_F}{\varepsilon - \varepsilon_0 + i\Gamma_F}} \right] \tag{1}$$

where T_0 is the background tunneling signal, A is the amplitude, q is the symmetry factor of the Frota function, ε_0 is the center of the resonance, and Γ_F is the Frota parameter (0.686 $\Gamma_F \sim k_B T_K$). These fits (shown in Figure 5E) reveal a steady decrease of T_K from 220 to 30 K as the tip is retracted from $z = 1.0$ to 2.5 nm.[47,48] This decrease can be explained by reduced screening of the unpaired electron spin by the underlying Au substrate as the ribbon is lifted off the surface.[31]

The experimental observation of an unpaired electron in N$_2$-5-AGNR trimers explains the smaller tunneling decay constant β since a secondary tunneling path through the Kondo resonance contributes to the overall through-conductance in this case.[32] The presence of a partially filled state on the N$_2$-5-AGNR trimer likely arises from a charging effect of the underlying Au substrate. Figure 5F shows the DFT-calculated molecular frontier orbital energies of a N$_2$-5-AGNR dimer, trimer, and tetramer (here, E_F is derived from Figure 4L). Coincidentally, E_F is seen to align with the highest occupied molecular orbital (HOMO) energy level of the N$_2$-5-AGNR trimer (Supporting Information Figure S6). This energy correlation facilitates partial charge transfer from the N$_2$-5-AGNR trimer to the underlying Au substrate. Although there is some uncertainty in the precise location of E_F, the observed Kondo resonance is consistent with the orbital picture presented in Figure 5F. Here, the HOMO of the dimer is low enough in energy that it is doubly occupied and thereby nonmagnetic. The tetramer is a similarly closed shell. The alignment of a trimer molecular orbital with E_F, on the other hand, gives rise to a singly occupied molecular orbital (SOMO) that hosts an unpaired electron spin and is able to form a Kondo singlet due to exchange coupling with itinerant electrons in the Au(111).

◼ CONCLUSIONS

Fermi-level engineering has been shown to be a versatile tool for inducing pronounced changes in the band structure and topology of GNRs. Molecular building blocks can be induced to undergo regiocontrolled self-assembly to yield N$_2$-5-AGNRs featuring a superlattice of N atom dopants along only one of the extended armchair edges. First-principles DFT-LDA calculations suggest that N-doping induces a sizable shift in the Fermi energy (~1.0 eV) and is accompanied by a change of the band structure from a direct to an indirect gap semiconductor. Hole-doping associated with the interaction of N$_2$-5-AGNRs and the underlying Au(111) substrate only partially counteracts the effective shift of E_F seen in experiments. Scanning tunneling spectroscopy and lift-off charge transport independently corroborate the overall theoretical picture. This work paves the way toward the realization of atomically defined band alignments at molecular-scale heterojunction and metal−semiconductor interfaces, critical components for any future molecular-scale technology.

■ EXPERIMENTAL SECTION

Materials and Instrumentation. Unless otherwise stated, all manipulations of air- and/or moisture-sensitive compounds were carried out in oven-dried glassware under an atmosphere of Ar. All solvents and reagents were purchased from Alfa Aesar, Spectrum Chemicals, Acros Organics, TCI America, and Sigma-Aldrich and used as received unless otherwise noted. Organic solvents were dried by passing through a column of alumina and were degassed by vigorous bubbling of N_2 through the solvent for 20 min. Flash column chromatography was performed on SiliCycle silica gel (particle size 40−63 μm). Thin-layer chromatography was carried out using SiliCycle silica gel 60 Å F-254 precoated plates (0.25 mm thick) and visualized by UV absorption. All ^1H and ^{13}C NMR spectra were recorded on Bruker AV-600, MHz spectrometers, and are referenced to residual solvent peaks ($CDCl_3$ ^1H NMR = 7.26 ppm, ^{13}C NMR = 77.16 ppm). High-resolution ESI mass spectrometry was performed on a Finnigan LTQFT (Thermo) mass spectrometer in positive ionization mode. High-resolution EI mass spectrometry was performed on an AutoSpec Premier (Waters) mass spectrometer in positive ionization mode. X-ray crystallography was performed on a Rigaku XtaLAB P200 equipped with a MicroMax 007HF dual-source rotating anode and a Pilatus 200 K hybrid pixel array detector. Data were collected using Mo Kα (λ = 0.71073 Å) radiation. Crystals were held at 100 K throughout the collection using an Oxford Cryostream 700. Data collection was performed with CrysAlisPro.[49] Data were processed with CrysAlisPro and includes a multiscan absorption correction applied using the SCALE3 ABSPACK scaling algorithm within CrysAlisPro. Crystallographic data was solved with ShelXT, refined with ShelXL, and finalized in Olex1.5. 4,7-Diiodobenzo[c]-[1,2,5]thiadiazole (**2**) was prepared following literature procedures.[34]

5,8-Diiodoquinoxaline (3). A 250 mL two-neck round-bottom flask was charged with 4,7-diiodobenzo[c][1,2,5]thiadiazole **2** (1.30 g, 3.35 mmol) in EtOH (60 mL) and THF (20 mL). A reflux condenser was attached, and the reaction mixture was degassed by sparging with N_2 for 20 min. $NaBH_4$ (0.380 g, 10.1 mmol) was slowly added to the reaction mixture in small portions. $CoCl_2·6H_2O$ was added (0.008 g, 0.034 mmol), and the reaction mixture was stirred for 8 h at 78 °C. The reaction mixture was concentrated, diluted with H_2O (50 mL), and extracted with Et_2O (100 mL). The combined organic phases were washed with H_2O (50 mL) and saturated aqueous NaCl (100 mL), dried over $MgSO_4$, and concentrated on a rotary evaporator. Crude 3,6-diiodobenzene-1,2-diamine (1.21 g) was used without further purification. A 100 mL two-neck round-bottom flask was charged with 3,6-diiodobenzene-1,2-diamine (1.21 g, 3.36 mmol) in EtOH (50 mL). A reflux condenser was attached, and the reaction mixture was degassed by sparging with N_2 for 20 min. Glyoxal (610 μL, 4.20 mmol, 40 wt % solution in H_2O) was added dropwise, and the reaction mixture was stirred under a N_2 atmosphere for 24 h at 78 °C. The reaction mixture was concentrated, diluted with H_2O (100 mL), and extracted with $CHCl_3$ (200 mL). The combined organic phases were washed with H_2O (50 mL), saturated aqueous NaCl (100 mL), dried over $MgSO_4$, and concentrated on a rotary evaporator. Column chromatography (SiO_2; 8:1:1 hexane/CH_2Cl_2/EtOAc) yielded **3** (0.077 g, 0.202 mmol, 6%) as a yellow solid. ^1H NMR (600 MHz, $CDCl_3$) δ = 8.91 (s, 2H), 8.13 (s, 2H) ppm; ^{13}C {^1H} NMR (151 MHz, $CDCl_3$) δ = 146.7, 143.3, 141.4 ppm; HRMS (EI-TOF) m/z: $[C_8H_4I_2N_2]^+$ calcd $[C_8H_4I_2N_2]^+$ 381.8464; found, 381.8468.

2-(4-Bromonaphthalen-1-yl)-4,4,5,5-tetramethyl-1,3,2-dioxaborolane (4). A 250 mL two-neck round-bottom flask was charged under N_2 with 1,4-dibromonaphthalene (3.00 g, 10.5 mmol) in dry THF (80 mL). The reaction mixture was cooled to −78 °C and stirred for 20 min. n-BuLi (5.04 mL, 12.6 mmol, 2.5 M in hexanes) was added dropwise, and the reaction mixture was stirred for 1 h at −78 °C. 2-Isopropoxy-4,4,5,5-tetramethyl-1,3,2-dioxaborolane (3.00 mL, 14.7 mmol) was added dropwise, and the reaction mixture was stirred for 8 h at −78 °C. The reaction mixture was quenched by the addition of 2 M HCl (10 mL), diluted with H_2O (100 mL), and extracted with EtOAc (200 mL). The combined organic phases were washed with H_2O (100 mL) and saturated aqueous NaCl (100 mL), dried over Na_2SO_4, and concentrated on a rotary evaporator. Column chromatography (SiO_2; 9:1:0.01 hexane/CH_2Cl_2/EtOAc) yielded **4** (3.11 g, 9.34 mmol, 89%) as a colorless crystalline solid. ^1H NMR (600 MHz, $CDCl_3$) δ = 8.80−8.75 (m, 1H), 8.31−8.25 (m, 1H), 7.89 (d, J = 7.4 Hz, 1H), 7.79 (s, 1H), 7.61−7.55 (m, 2H), 1.42 (s, 13H) ppm; ^{13}C {^1H} NMR (151 MHz, $CDCl_3$) δ = 138.3, 135.8, 131.9, 129.5, 129.0, 127.5, 127.4, 127.3, 127.1, 84.1, 25.1 ppm; HRMS (EI-TOF) m/z: $[C_{16}H_{18}BBrO_2]^+$ calcd $[C_{16}H_{18}BBrO_2]$ 334.0563; found, 334.0563.

5,8-Bis(4-bromonaphthalen-1-yl)quinoxaline (1a). A 25 mL Schlenk tube was charged with **3** (0.050 g, 0.131 mmol), **4** (0.089 g, 0.269 mmol), bis(triphenylphosphine)palladium(II) dichloride (0.009 g, 0.013 mmol), and Na_2CO_3 (0.056 g, 0.524 mmol) in dioxane (5 mL) and H_2O (2 mL). The reaction mixture was degassed by sparging with N_2 for 20 min and stirred under N_2 for 18 h at 80 °C. The reaction mixture was diluted with H_2O (20 mL) and extracted with CH_2Cl_2 (70 mL). The combined organic phases were washed with H_2O (50 mL) and saturated aqueous NaCl (50 mL), dried over Na_2SO_4, and concentrated on a rotary evaporator. Column chromatography (SiO_2; 1:1:0.1 hexane/CH_2Cl_2/EtOAc) yielded **1a** (0.056 g, 0.103 mmol, 79%) as a yellow solid. ^1H NMR (600 MHz, $CDCl_3$) δ = 8.73 (s, 2H), 8.45−8.32 (m, 2H), 7.97 (t, J = 7.7 Hz, 2H), 7.93 (s, 2H), 7.63 (dddd, J = 11.3, 8.3, 6.6, 1.3 Hz, 2H), 7.55−7.52 (m, 2H), 7.49 (d, J = 7.5 Hz, 2H), 7.47−7.39 (m, 4H) ppm; ^{13}C {^1H} NMR (151 MHz, $CDCl_3$) δ = 145.1, 145.1, 142.1, 142.1, 139.9, 139.9, 136.8, 136.8, 134.1, 134.1, 132.2, 132.2, 131.6, 131.5, 129.6, 129.6, 128.7, 128.6, 127.8, 127.8, 127.4, 127.4, 127.1, 127.1, 127.0, 127.0, 123.7, 123.6 ppm; HRMS (EI-TOF) m/z: $[C_{28}H_{16}Br_2N_2]^+$ calcd $[C_{28}H_{16}Br_2N_2]$ 539.9660; found 539.9656.

5,8-Bis(4-iodonaphthalen-1-yl)quinoxaline (1b). A 10 mL Schlenk tube was charged under N_2 with **1a** (0.042 g, 0.078 mmol), NaI (0.089 g, 0.777 mmol), CuI (0.009 g, 0.016 mmol), and racemic trans-N,N′-dimethyl-1,2-cyclohexanediamine (2.5 μL, 0.02 mmol) in dry, degassed dioxane (2 mL). The Schlenk tube was sealed with a Teflon valve, and the reaction mixture was stirred under N_2 for 24 h at 110 °C. The reaction mixture was diluted with H_2O (10 mL) and extracted with EtOAc (30 mL). The combined organic phases were washed with H_2O (10 mL) and saturated aqueous NaCl (10 mL), dried over Na_2SO_4, and concentrated on a rotary evaporator. Column chromatography (SiO_2; 9:1:0.2 hexane/CH_2Cl_2/EtOAc) yielded **1b** (0.018 g, 0.029 mmol, 37%) as a colorless crystalline solid. ^1H NMR (600 MHz, $CDCl_3$) δ = 8.73 (s, 2H), 8.26 (ddd, J = 12.2, 8.0, 4.9 Hz, 4H), 7.92 (s, 2H), 7.65−7.54 (m, 2H), 7.54−7.28 (m, 6H) ppm; ^{13}C {^1H} NMR (151 MHz, $CDCl_3$) δ = 145.1, 145.1, 142.1, 139.9, 139.9, 137.1, 137.1, 134.4, 134.4, 133.6, 132.9, 131.5, 129.3, 129.2, 127.9, 127.8, 127.2, 127.2, 127.1, 127.0, 100.6 ppm; HRMS (ESI-TOF) m/z: $[C_{28}H_{16}I_2N_2]^+$ calcd $[C_{28}H_{16}I_2N_2]$ 634.9476; found, 634.9470.

N_2-5-AGNR Growth on Au(111) Surfaces. N_2-5-AGNRs were grown on Au(111)/mica films under UHV conditions. Atomically clean Au(111) surfaces were prepared through iterative Ar$^+$ sputter/anneal cycles. Submonolayer coverage of **1b** on atomically clean Au(111) was obtained by sublimation at crucible temperatures of 453−473 K using a Knudsen cell evaporator. After deposition, the surface temperature was slowly ramped (≤2 K min^{-1}) to 453 K and held at this temperature for 15 min to induce the radical step growth polymerization and then slowly ramped (≤2 K min^{-1}) to 623 K and held there for 15 min to induce cyclodehydrogenation.

Scanning Tunneling Microscopy and Spectroscopy. All STM experiments were performed using a commercial Createc LT-STM operating at T = 4 K and using tungsten STM tips. dI/dV measurements were recorded with CO-functionalized STM tips using a lock-in amplifier with a modulation frequency of f = 533 Hz and a modulation amplitude of V_{ac} = 1.0−4.0 mV. dI/dV point spectra and dI/dV maps were recorded in the constant height mode.

Calculations. First-principles DFT calculations in the LDA and LSDA approximations were implemented using the Quantum Espresso package.[50] We used norm-conserving (NC) pseudopotentials with a 60 Ry energy cutoff and 0.005 Ry Gaussian broadening.

To ensure the accuracy of our results, a sufficiently large vacuum region was included in the supercell calculation. All of the dangling bonds at the edges of the carbon skeleton were hydrogenated. The structures were fully relaxed until all components of the force were smaller than 0.01 eV/ Å. A periodic infinite model of N_2-5-AGNR was used to calculate band structures in Figures 1B−C and 4L, and LDOS maps in Figure 4G−I, K. A finite-length model of a N_2-5-AGNR was used to calculate the LDOS maps in Figure 4J and energy levels in Figure 5F.

ASSOCIATED CONTENT

Ⓢ Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/jacs.3c05755.

> Additional experimental details, STM topographic images, STS lift-off experiments, computational data, materials and methods, all NMR spectra, and X-ray crystal structure data for **1b** (PDF)

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

REFERENCES

(1) Cai, J. M.; Ruffieux, P.; Jaafar, R.; Bieri, M.; Braun, T.; Blankenburg, S.; Muoth, M.; Seitsonen, A. P.; Saleh, M.; Feng, X. L.; Müllen, K.; Fasel, R. Atomically precise bottom-up fabrication of graphene nanoribbons. *Nature* **2010**, *466*, 470−473.

(2) Rizzo, D. J.; Veber, G.; Jiang, J.; McCurdy, R.; Cao, T.; Bronner, C.; Chen, T.; Louie, S. G.; Fischer, F. R.; Crommie, M. F. Inducing metallicity in graphene nanoribbons via zero-mode superlattices. *Science* **2020**, *369*, 1597−1603.

(3) Mishra, S.; Catarina, G.; Wu, F. P.; Ortiz, R.; Jacob, D.; Eimre, K.; Ma, J.; Pignedoli, C. A.; Feng, X. L.; Ruffieux, P.; Fernandez-Rossier, J.; Fasel, R. Observation of fractional edge excitations in nanographene spin chains. *Nature* **2021**, *598*, 287−292.

(4) Cao, T.; Zhao, F. Z.; Louie, S. G. Topological Phases in Graphene Nanoribbons: Junction States, Spin Centers, and Quantum Spin Chains. *Phys. Rev. Lett.* **2017**, *119*, No. 076401.

(5) Rizzo, D. J.; Veber, G.; Cao, T.; Bronner, C.; Chen, T.; Zhao, F. Z.; Rodriguez, H.; Louie, S. G.; Crommie, M. F.; Fischer, F. R. Topological band engineering of graphene nanoribbons. *Nature* **2018**, *560*, 204−208.

(6) Gröning, O.; Wang, S. Y.; Yao, X. L.; Pignedoli, C. A.; Barin, G. B.; Daniels, C.; Cupo, A.; Meunier, V.; Feng, X. L.; Narita, A.; Mullen, K.; Ruffieux, P.; Fasel, R. Engineering of robust topological quantum phases in graphene nanoribbons. *Nature* **2018**, *560*, 209−213.

(7) Jiang, J.; Louie, S. G. Topology Classification using Chiral Symmetry and Spin Correlations in Graphene Nanoribbons. *Nano Lett.* **2021**, *21*, 197−202.

(8) Son, Y. W.; Cohen, M. L.; Louie, S. G. Energy gaps in graphene nanoribbons. *Phys. Rev. Lett.* **2006**, *97*, No. 216803.

(9) Chen, Y. C.; de Oteyza, D. G.; Pedramrazi, Z.; Chen, C.; Fischer, F. R.; Crommie, M. F. Tuning the Band Gap of Graphene Nanoribbons Synthesized from Molecular Precursors. *ACS Nano* **2013**, *7*, 6123−6128.

(10) Durr, R. A.; Haberer, D.; Lee, Y. L.; Blackwell, R.; Kalayjian, A. M.; Marangoni, T.; Ihm, J.; Louie, S. G.; Fischer, F. R. Orbitally Matched Edge-Doping in Graphene Nanoribbons. *J. Am. Chem. Soc.* **2018**, *140*, 807−813.

(11) Jacobse, P. H.; Jin, Z. X.; Jiang, J. W.; Peurifoy, S.; Yue, Z. Q.; Wang, Z. Y.; Rizzo, D. J.; Louie, S. G.; Nuckolls, C.; Crommie, M. F. Pseudo-atomic orbital behavior in graphene nanoribbons with four-membered rings. *Sci. Adv.* **2021**, *7*, No. eabl5892.

(12) Sánchez-Sánchez, C.; Dienel, T.; Nicola, A.; Kharche, N.; Liang, L. B.; Daniels, C.; Meunier, V.; Liu, J. Z.; Feng, X. L.; Mllen, K.; Sanchez-Valencia, J. R.; Grning, O.; Ruffieux, P.; Fasel, R. On-Surface Synthesis and Characterization of Acene-Based Nanoribbons Incorporating Four-Membered Rings. *Chem.— Eur. J.* **2019**, *25*, 12074−12082.

(13) Osella, S.; Narita, A.; Schwab, M. G.; Hernandez, Y.; Feng, X. L.; Mullen, K.; Beljonne, D. Graphene Nanoribbons as Low Band Gap Donor Materials for Organic Photovoltaics: Quantum Chemical Aided Design. *ACS Nano* **2012**, *6*, 5539−5548.

(14) Villegas, C. E. P.; Mendonca, P. B.; Rocha, A. R. Optical spectrum of bottom-up graphene nanoribbons: towards efficient atom-thick excitonic solar cells. *Sci. Rep.* **2014**, *4*, 6579.

(15) Ye, Y.; Gan, L.; Dai, L.; Meng, H.; Wei, F.; Dai, Y.; Shi, Z. J.; Yu, B.; Guo, X. F.; Qin, G. G. Multicolor graphene nanoribbon/ semiconductor nanowire heterojunction light-emitting diodes. *J. Mater. Chem.* **2011**, *21*, 11760−11763.

(16) Junaid, M.; Khir, M. H. M.; Witjaksono, G.; Ullah, Z.; Tansu, N.; Saheed, M. S. M.; Kumar, P.; Wah, L. H.; Magsi, S. A.; Siddiqui, M. A. A Review on Graphene-Based Light Emitting Functional Devices. *Molecules* **2020**, *25*, 4217.

(17) Palacios-Berraquero, C.; Barbone, M.; Kara, D. M.; Chen, X. L.; Goykhman, I.; Yoon, D.; Ott, A. K.; Beitner, J.; Watanabe, K.; Taniguchi, T.; Ferrari, A. C.; Atature, M. Atomically thin quantum light-emitting diodes. *Nat. Commun.* **2016**, *7*, No. 12978.

(18) Miao, W. J.; Wang, L.; Mu, X. J.; Wang, J. G. The magical photoelectric and optoelectronic properties of graphene nanoribbons and their applications. *J. Mater. Chem. C* **2021**, *9*, 13600−13616.

(19) Tavousi, A.; Mansouri-Birjandi, M. A.; Janfaza, M. Optoelectronic application of graphene nanoribbon for mid-infrared bandpass filtering. *Appl. Opt.* **2018**, *57*, 5800−5805.

(20) Cai, J. M.; Pignedoli, C. A.; Talirz, L.; Ruffieux, P.; Sode, H.; Liang, L. B.; Meunier, V.; Berger, R.; Li, R. J.; Feng, X. L.; Müllen, K.; Fasel, R. Graphene nanoribbon heterojunctions. *Nat. Nanotechnol.* **2014**, *9*, 896−900.

(21) Bronner, C.; Stremlau, S.; Gille, M.; Brausse, F.; Haase, A.; Hecht, S.; Tegeder, P. Aligning the Band Gap of Graphene Nanoribbons by Monomer Doping. *Angew. Chem., Int. Ed.* **2013**, *52*, 4422−4425.

(22) Vo, T. H.; Shekhirev, M.; Kunkel, D. A.; Orange, F.; Guinel, M. J. F.; Enders, A.; Sinitskii, A. Bottom-up solution synthesis of narrow nitrogen-doped graphene nanoribbons. *Chem. Commun.* **2014**, *50*, 4172−4174.

(23) Zhang, Y.; Zhang, Y. F.; Li, G.; Lu, J. C.; Lin, X.; Du, S. X.; Berger, R.; Feng, X. L.; Mullen, K.; Gao, H. J. Direct visualization of atomically precise nitrogen-doped graphene nanoribbons. *Appl. Phys. Lett.* **2014**, *105*, No. 023101.

(24) Vo, T. H.; Perera, U. G. E.; Shekhirev, M.; Pour, M. M.; Kunkel, D. A.; Lu, H. D.; Gruverman, A.; Sutter, E.; Cotlet, M.; Nykypanchuk, D.; Zahl, P.; Enders, A.; Sinitskii, A.; Sutter, P. Nitrogen-Doping Induced Self-Assembly of Graphene Nanoribbon-Based Two-Dimensional and Three-Dimensional Metamaterials. *Nano Lett.* **2015**, *15*, 5770−5777.

(25) Marangoni, T.; Haberer, D.; Rizzo, D. J.; Cloke, R. R.; Fischer, F. R. Heterostructures through Divergent Edge Reconstruction in Nitrogen-Doped Segmented Graphene Nanoribbons. *Chem. − Eur. J.* **2016**, *22*, 13037−13040.

(26) Rizzo, D. J.; Wu, M.; Tsai, H. Z.; Marangoni, T.; Durr, R. A.; Omrani, A. A.; Liou, F.; Bronner, C.; Joshi, T.; Nguyen, G. D.; Rodgers, G. F.; Choi, W. W.; Jorgensen, J. H.; Fischer, F. R.; Louie, S. G.; Crommie, M. F. Length-Dependent Evolution of Type II Heterojunctions in Bottom-Up-Synthesized Graphene Nanoribbons. *Nano Lett.* **2019**, *19*, 3221−3228.

(27) Pawlak, R.; Liu, X. S.; Ninova, S.; D'Astolfo, P.; Drechsel, C.; Sangtarash, S.; Haner, R.; Decurtins, S.; Sadeghi, H.; Lambert, C. J.; Aschauer, U.; Liu, S. X.; Meyer, E. Bottom-up Synthesis of Nitrogen-Doped Porous Graphene Nanoribbons. *J. Am. Chem. Soc.* **2020**, *142*, 12568−12573.

(28) Li, J. C.; Brandimarte, P.; Vilas-Varela, M.; Merino-Diez, N.; Moreno, C.; Mugarza, A.; Mollejo, J. S.; Sanchez-Portal, D.; de Oteyza, D. G.; Corso, M.; Garcia-Lekue, A.; Pena, D.; Pascual, J. I. Band Depopulation of Graphene Nanoribbons Induced by Chemical Gating with Amino Groups. *ACS Nano* **2020**, *14*, 1895−1901.

(29) Blackwell, R. E.; Zhao, F. Z.; Brooks, E.; Zhu, J. M.; Piskun, I.; Wang, S. K.; Delgado, A.; Lee, Y. L.; Louie, S. G.; Fischer, F. R. Spin splitting of dopant edge state in magnetic zigzag graphene nanoribbons. *Nature* **2021**, *600*, 647−652.

(30) Chang, X.; Huang, L.; Gao, H. Y.; Fu, Y.; Ma, J.; Yang, H.; Liu, J.; Fu, X.; Lin, X.; Feng, X.; Du, S. X.; Gao, H. On-surface synthesis and edge states of NBN-doped zigzag graphene nanoribbons. *Nano Res.* **2023**, *16*, 10436−10442, DOI: 10.1007/s12274-023-5605-2.

(31) Wen, E. C. H.; Jacobse, P. H.; Jiang, J.; Wang, Z. Y.; McCurdy, R. D.; Louie, S. G.; Crommie, M. F.; Fischer, F. R. Magnetic Interactions in Substitutional Core-Doped Graphene Nanoribbons. *J. Am. Chem. Soc.* **2022**, *144*, 13696−13703.

(32) Lawrence, J.; Brandimarte, P.; Berdonces-Layunta, A.; Mohammed, M. S. G.; Grewal, A.; Leon, C. C.; Sanchez-Portal, D.; de Oteyza, D. G. Probing the Magnetism of Topological End States in 5-Armchair Graphene Nanoribbons. *ACS Nano* **2020**, *14*, 4499−4508.

(33) Zak, J. Berry's Phase for Energy-Bands in Solids. *Phys. Rev. Lett.* **1989**, *62*, 2747−2750.

(34) Shimada, M.; Tsuchiya, M.; Sakamoto, R.; Yamanoi, Y.; Nishibori, E.; Sugimoto, K.; Nishihara, H. Bright Solid-State Emission of Disilane-Bridged Donor-Acceptor-Donor and Acceptor-Donor-Acceptor Chromophores. *Angew. Chem., Int. Ed.* **2016**, *55*, 3022−3026.

(35) Bronner, C.; Marangoni, T.; Rizzo, D. J.; Durr, R. A.; Jorgensen, J. H.; Fischer, F. R.; Crommie, M. F. Iodine versus Bromine Functionalization for Bottom-Up Graphene Nanoribbon Growth: Role of Diffusion. *J. Phys. Chem. C* **2017**, *121*, 18490−18495.

(36) Piskun, I.; Blackwell, R.; Jornet-Somoza, J.; Zhao, F. Z.; Rubio, A.; Louie, S. G.; Fischer, F. R. Covalent C-N Bond Formation through a Surface Catalyzed Thermal Cyclodehydrogenation. *J. Am. Chem. Soc.* **2020**, *142*, 3696−3700.

(37) Wang, X. Y.; Richter, M.; He, Y. Q.; Bjork, J.; Riss, A.; Rajesh, R.; Garnica, M.; Hennersdorf, F.; Weigand, J. J.; Narita, A.; Berger, R.; Feng, X. L.; Auwarter, W.; Barth, J. V.; Palma, C. A.; Mullen, K. Exploration of pyrazine-embedded antiaromatic polycyclic hydro-carbons generated by solution and on-surface azomethine ylide homocoupling. *Nat. Commun.* **2017**, *8*, No. 1948.

(38) Koch, M.; Ample, F.; Joachim, C.; Grill, L. Voltage-dependent conductance of a single graphene nanoribbon. *Nat. Nanotechnol.* **2012**, *7*, 713−717.

(39) Jacobse, P. H.; Mangnus, M. J. J.; Zevenhuizen, S. J. M.; Swart, I. Mapping the Conductance of Electronically Decoupled Graphene Nanoribbons. *ACS Nano* **2018**, *12*, 7048−7056.

(40) Friedrich, N.; Brandimarte, P.; Li, J. C.; Saito, S.; Yamaguchi, S.; Pozo, I.; Pena, D.; Frederiksen, T.; Garcia-Lekue, A.; Sanchez-Portal, D.; Pascual, J. I. Magnetism of Topological Boundary States

Induced by Boron Substitution in Graphene Nanoribbons. *Phys. Rev. Lett.* **2020**, *125*, No. 146801.

(41) Jacobse, P. H.; Kimouche, A.; Gebraad, T.; Ervasti, M. M.; Thijssen, M.; Liljeroth, P.; Swart, I. Electronic components embedded in a single graphene nanoribbon. *Nat. Commun.* **2017**, *8*, No. 119.

(42) Kondo, J. Effect of Ordinary Scattering on Exchange Scattering from Magnetic Impurity in Metals. *Phys. Rev.* **1968**, *169*, 437−440.

(43) Madhavan, V.; Chen, W.; Jamneala, T.; Crommie, M. F.; Wingreen, N. S. Tunneling into a single magnetic atom: Spectroscopic evidence of the Kondo resonance. *Science* **1998**, *280*, 567−569.

(44) Li, J. T.; Schneider, W. D.; Berndt, R.; Delley, B. Kondo scattering observed at a single magnetic impurity. *Phys. Rev. Lett.* **1998**, *80*, 2893−2896.

(45) Li, J. C.; Sanz, S.; Corso, M.; Choi, D. J.; Pena, D.; Frederiksen, T.; Pascual, J. I. Single spin localization and manipulation in graphene open-shell nanostructures. *Nat. Commun.* **2019**, *10*, No. 200.

(46) Sun, Q.; Mateo, L. M.; Robles, R.; Lorente, N.; Ruffieux, P.; Bottari, G.; Torres, T.; Fasel, R. Bottom-up Fabrication and Atomic-Scale Characterization of Triply Linked, Laterally π-Extended Porphyrin Nanotapes. *Angew. Chem., Int. Ed.* **2021**, *60*, 16208−16214.

(47) Frank, S.; Jacob, D. Orbital signatures of Fano-Kondo line shapes in STM adatom spectroscopy. *Phys. Rev. B* **2015**, *92*, No. 235127.

(48) Frota, H. O. Shape of the Kondo Resonance. *Phys. Rev. B* **1992**, *45*, 1096−1099.

(49) *CrysAlisPro, 1.171.40.82a, 1.171.41.116a*; Oxford Diffraction/Agilent Technologies UK Ltd.: Oxford, UK, 2015.

(50) Giannozzi, P.; Baroni, S.; Bonini, N.; Calandra, M.; Car, R.; Cavazzoni, C.; Ceresoli, D.; Chiarotti, G. L.; Cococcioni, M.; Dabo, I.; Dal Corso, A.; de Gironcoli, S.; Fabris, S.; Fratesi, G.; Gebauer, R.; Gerstmann, U.; Gougoussis, C.; Kokalj, A.; Lazzeri, M.; Martin-Samos, L.; Marzari, N.; Mauri, F.; Mazzarello, R.; Paolini, S.; Pasquarello, A.; Paulatto, L.; Sbraccia, C.; Scandolo, S.; Sclauzero, G.; Seitsonen, A. P.; Smogunov, A.; Umari, P.; Wentzcovitch, R. M. QUANTUM ESPRESSO: a modular and open-source software project for quantum simulations of materials. *J. Phys.: Condens. Mater.* **2009**, *21*, No. 395502.

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