

Panchromatic Ternary Photovoltaic Cells Using a Non-Fullerene Acceptor Synthesized Using C-H Functionalization

Junxiang Zhang,^{†,‡} Cenqi Yan,^{‡,§} Wei Wang,[‡] Yiqun Xiao,[‡] Xinhui Lu,[‡] Stephen Barlow,[†] Timothy C. Parker,[†] Xiaowei Zhan^{‡,*} and Seth R. Marder^{†,*}

[†]Center for Organic Photonics and Electronics, and School of Chemistry and Biochemistry, Georgia Institute of Technology, Atlanta, GA 30332-0400, US

[‡]Department of Materials Science and Engineering, College of Engineering, Key Laboratory of Polymer Chemistry and Physics of Ministry of Education, Peking University, Beijing 100871, China

[§]Department of Physics, The Chinese University of Hong Kong, New Territories, Hong Kong, China

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ABSTRACT: A new non-fullerene acceptor (IHIC-N) differs from IDT acceptors in having a naphthalene rather than a benzene ring in the core. A key intermediate in the synthesis of IHIC-N, which has a quasi-linear 2,3,6,7-substituted core rather than the 1,2,5,6-pattern of recently reported regioisomers accessed through electrophilic substitution, is dimethyl 3,7-bis(2-thienyl)naphthalene-2,6-dicarboxylate, previously obtained in six steps from 2,6-dihydroxynaphthalene, but here obtained from naphthalene-2,6-dicarboxylic acid in 62% yield using palladium-catalyzed carboxylate-directed C-H bond functionalization. IHIC-N exhibits an absorption maximum of 641 nm in chloroform. Solar cells with PTB7-Th as the donor give a power conversion efficiency of 6.91%. However, the new acceptor was more effectively used in a ternary blend with PTB7-Th and IHIC, a related non-fullerene acceptor based on a thienothiophene core. The best device was obtained for a 1:0.5:1 PTB7-Th:IHIC-N:IHIC blend and gave an open-circuit voltage of 0.785 V, a short-circuit current density of 21.3 mA cm⁻², a fill factor of 70.8%, and a power conversion efficiency of 11.9%. The complementarity of IHIC-N and IHIC absorption, balanced carrier mobilities, and appropriate domain sizes for effective exciton harvesting likely all contribute to one of the highest efficiencies obtained for a single-junction fullerene-free PTB7-Th device.

Some of the most significant recent advances in organic solar cells (OSCs) have been in developing non-fullerene acceptors (NFAs),¹⁻⁷ potential advantages of which over fullerenes include easier and lower cost synthesis and purification, stronger absorption, and easier tunability of absorption bands and electron affinities (EAs) via structural changes.^{8,9} Indacenodithiophene (IDT)-based NFAs have attracted particular interest, owing to their high electron mobilities (comparable to or exceeding those of fullerenes) and strong absorption at ca. 600-800 nm.^{10,11} To complement their absorption in bulk heterojunction blends, wide- or medium-bandgap polymers are usually chosen as donors.¹² However, to exploit the low-bandgap donor polymers extensively developed for use with fullerenes, NFAs with even lower bandgaps in the near-infrared (NIR), are desirable. IHIC (Figure 1),¹³ in which the central arene of IDT derivatives is replaced with thienothiophene, was recently reported as a NIR-absorbing NFA. OSCs based on binary blends of IHIC with PTB7-Th (Figure 1) show high power conversion efficiency (PCE) of 9.77%.

To increase the PCE of PTB7-Th:IHIC OSCs, we were inspired by recent reports of OSCs using ternary blends of a

donor polymer with two acceptors.¹⁴⁻¹⁷ The third component should ideally: (i) have an EA lower than that of IHIC, in order to avoid compromising V_{OC} , but still sufficiently large to accept an electron from photoexcited PTB7-Th; and (ii) absorb at higher energy than IHIC and PTB7-Th and/or significantly more strongly than PTB7-Th in order to afford a more panchromatic absorption and higher short-circuit current density (J_{SC}). Here we report a new NFA – IHIC-N (Figure 1) – in which a naphthalene core is chosen to provide a higher π -surface area than that of IDT to potentially interact more strongly with the polymer π -system. The synthesis of a key intermediate is enabled by a double carboxylate-directed C-H bond functionalization.¹⁸ Optimized ternary PTB7-Th:IHIC-N:IHIC OSCs give PCE = 11.9%.

Molecules similar to IHIC-N (structures **I** and **II**, Figure 1) have recently been reported,^{19,20} however, these differ in the solubilizing substituents (R) and in the regiosomerism of the naphthalene cores, which are 1,2,5,6-substituted (rather than the quasi-linear 2,3,6,7-substitution pattern in IHIC-N). The synthesis of both of these cores depends on the α -positions (C-1 and C-5) of naphthalenes generally

being the most active towards electrophilic reactions.²¹ “S”-shaped compounds of type I ultimately rely on the electrophilic 1,5-dibromination of 2,6-dimethylnaphthalene, while the cores of compounds of type II are synthesized through intramolecular Friedel-Crafts alkylation/cyclizations on to the 1,5-positions of 2,6-disubstituted naphthalenes. Compounds of type II ultimately rely on the electrophilic 1,5-dibromination of 2,6-dimethylnaphthalene. Our synthesis of IHIC-N relies on the availability of **1** (Figure 2), the recently reported synthesis of which requires six steps with an overall 16% yield.²² We developed a much more atom-economical approach to **1** using a highly regioselective double C-H functionalization of 2,6-naphthalic acid. Carboxylate-directed bromination conditions first developed by Yu²³ gave 3,7-dibromo-2,6-naphthalic acid, which was then esterified, and coupled with 2-(tributylstannyl)thiophene to give **1** in 62% overall yield. The regiochemistry of the dibromination, and therefore of **1**, was confirmed using 2D NMR methods (see Supporting Information). Bromination adjacent to the carboxylic acid is attributable to chelation to form five-membered palladacycle intermediates,²³ while 1,5-dibromination is presumably precluded by steric interactions of the Pd with the 4,8-hydrogen atoms (Figure 2c,d).

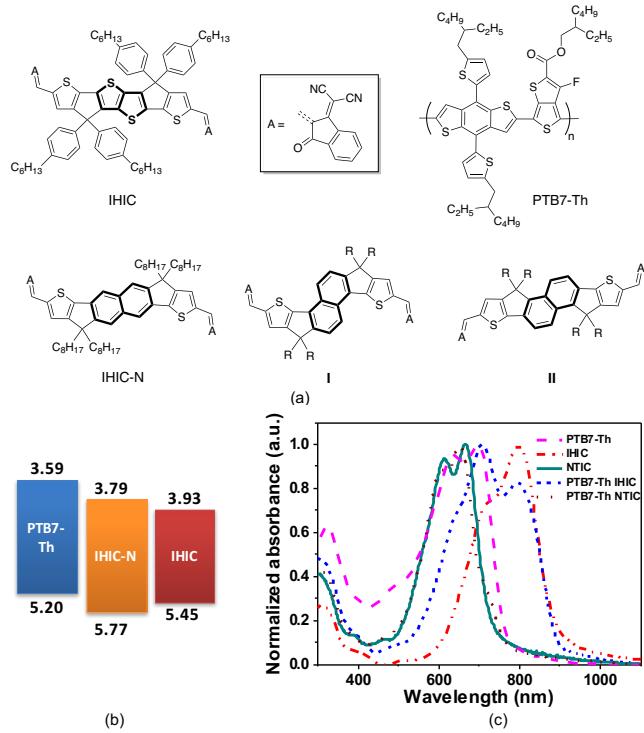


Figure 1. (a) Chemical structures of materials discussed in the text. (b) Estimated IEs and EAs for PTB7-Th, IHIC-N, and IHIC. (c) Thin-film UV-Vis-NIR absorption spectra of PTB7-Th, IHIC, IHIC-N, and blends.

IHIC-N exhibits an absorption band with $\epsilon_{\text{max}} = 1.2 \times 10^5 \text{ M}^{-1} \text{ cm}^{-1}$ (ca. $\frac{2}{3}$ that of IHIC) at $\lambda_{\text{max}} = 641 \text{ nm}$ in CHCl_3 (Figure S3). The absorption maximum is similar to that for **I** ($R = \text{alkyl}$)¹⁹ and is blue-shifted slightly (0.04 eV) and significantly (0.28 eV) relative to a core-alkylated IDT derivative⁴ and to IHIC, respectively.¹³ The blue shift vs. IHIC is also

seen in thin-film absorption spectra in neat films and blends with PTB7-Th (Figure 1c). This difference can be mainly attributed to disruption of the delocalization of the frontier molecular orbitals in IHIC-N, relative to those of IHIC, by the more aromatic central part of the core (Figure S5). This localization is also reflected in the frontier energy levels (Figure 1b) estimated from cyclic voltammetry (see Supporting Information), in which IHIC-N exhibits higher ionization energy (IE, by 0.13 eV) and lower EA (by 0.12 eV) than IHIC.

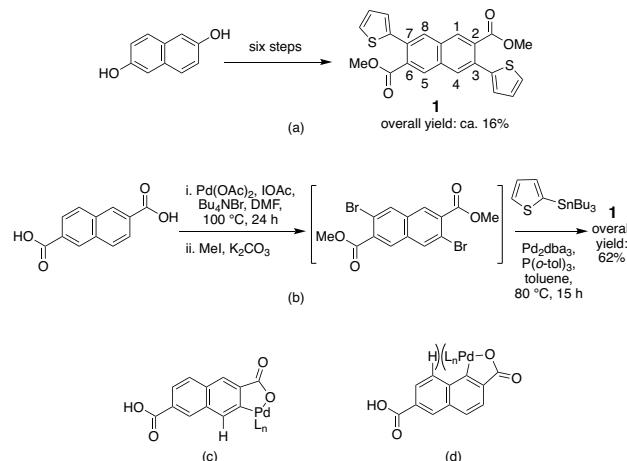


Figure 2. (a) Six-step synthesis of key intermediate **1**.²² (b) Synthesis of **1** based on carboxylic acid-directed 3,7-dibromination and cross coupling. (c) Presumed intermediate in 3-bromination and (d) intermediate for 1-bromination, destabilized by Pd/8-hydrogen steric interactions.

There is considerable overlap between the photoluminescence spectrum of IHIC-N and the absorption spectrum of the IHIC, indicating the possibility of energy transfer from the former to the latter (Figure S4a). The emission spectra of films of neat IHIC-N, neat IHIC, and a 1:2 w/w IHIC-N:IHIC blend, all excited at 305 nm, were also compared (Figure S4b). The emission of neat IHIC-N is relatively strong and broad (at ca. 650–850 nm) and that of IHIC (at ca. 770–850 nm) is relatively weak. The emission of IHIC-N is strongly quenched in the blend, and that of IHIC somewhat enhanced, consistent with efficient energy transfer from IHIC-N to IHIC.

ITO/ZnO/PTB7-Th:IHIC-N:IHIC/MoO₃/Ag OSCs were fabricated using a PTB7-Th:(total NFA) ratio of 1:1.5 (w/w) and varying the NFA from 100% IHIC-N to 100% IHIC (Table 1). Fabrication conditions, such as additives and spin-coating speed, were optimized for PTB7-Th:IHIC-N (Table S1). 1,8-Diiodooctane (DIO) is known to improve the performance of PTB7-Th:IHIC,²⁴ we found it also improves PTB7-Th:IHIC-N OSCs, reducing average J_{SC} , but increasing the open-circuit voltage (V_{OC}) and fill factor (FF), resulting in slightly higher average PCE. All parameters are strongly affected by the addition of IHIC. With increasing IHIC content, average V_{OC} gradually decreases, although not in a linear fashion (Figure S7); presumably the higher EA of IHIC plays a role. The average FF generally increases. J_{SC} and PCE are both maximized at a 1:2 ratio and the PCE of 11.9% for these devices is among the highest for single-

junction fullerene-free PTB7-Th OSCs (Figure 3a). However, the PCE of the ternary OSCs is relatively insensitive to the NFA composition, exceeding 11% for IHIC-N:IHIC ratios varying from 1:1 to 1:5.

EQE spectra (Figures 3b, S6, and S8) show a clear enhancement for the champion ternary device over the PTB7-Th:IHIC-N device, accounting for the improved J_{SC} ; in particular, the high EQE values seen at ca. 750–850 nm are attributable to the strong NIR absorption of IHIC. The most obvious improvement in the EQE of the champion device over that of a PTB7-Th:IHIC control is at 500–700 nm, where IHIC-N absorbs. IHIC-N excitons may, in principle at least, be harvested both by electron transfer from PTB7-Th and by energy transfer to IHIC (see above), followed by electron transfer from the donor. However, the champion ternary blend also shows somewhat improved EQE at ca. 750–850 nm relative to the control, despite absorbing less strongly in this range (Figure S8).

The space-charge limited current (SCLC) method was used to measure hole and electron mobilities (μ_h and μ_e) of blend films (Figures S9 and S10; Table S2). A ternary blend (1:2 IHIC-N:IHIC) exhibited $\mu_h = 1.1 \times 10^{-3} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, somewhat higher than either PTB7-Th:IHIC-N ($4.3 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$) or PTB7-Th:IHIC ($7.1 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$). Values of μ_e show more variation: 2.8×10^{-5} , 1.6×10^{-3} and $3.1 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for PTB7-Th:IHIC-N, PTB7-Th:IHIC, PTB7-Th:IHIC-N:IHIC, respectively.²⁵ These data suggest that the ternary blends and PTB7-Th:IHIC blends both exhibit more balanced charge transport than PTB7-Th:IHIC-N blends, consistent with the much lower FF seen for the IHIC-free devices (Table 1).

Table 1. Data for PTB7:Acceptor (1:1.5 w/w) OSCs with Different IHIC-N:IHIC Ratios as Acceptor^a

Acceptor	%v/v DIO ^b	V_{OC} / mV	J_{SC} / mA cm ⁻²	Calc. J_{SC} ^c / mA cm ⁻²	FF / %	PCE / %
IHIC-N	0	832 (832±6)	14.1 (13.8±0.4)	14.3	57.5 (56.1±1.0)	6.74 (6.46±0.17)
IHIC-N	0.5	846 (841±3)	13.5 (13.4±0.4)	13.3	60.6 (58.6±1.0)	6.91 (6.59±0.18)
2:1 IHIC-N:IHIC	0.5	807 (807±2)	18.8 (18.8±0.3)	18.0	69.3 (66.0±1.7)	10.5 (10.0±0.3)
1:1 IHIC-N:IHIC	0.5	787 (790±5)	20.5 (20.4±0.3)	19.7	68.8 (67.1±1.2)	11.1 (10.9±0.2)
1:2 IHIC-N:IHIC	0.5	785 (782±2)	21.3 (21.2±0.3)	20.3	70.8 (68.8±1.2)	11.9 (11.5±0.3)
1:5 IHIC-N:IHIC	0.5	781 (778±4)	20.1 (19.9±0.4)	19.8	70.3 (68.7±1.7)	11.0 (10.6±0.3)
IHIC	0.5	752 (749±8)	19.4 (19.4±0.3)	18.5	72.5 (70.0±1.8)	10.6 (10.2±0.3)

^aValues are for the highest-PCE device, with average values and standard deviations obtained from 20 devices listed in parentheses.

^bDIO = 1,8-diiodooctane. ^cExpected J_{SC} value from integration of the EQE spectra with the AM 1.5G reference spectrum; these values are within 5% of those from J - V curves.

GIWAXS (Figures 4a-d and S12) and GISAXS (Figures 4e and S13) were used to investigate molecular packing and nanoscale phase separation²⁶ in thin films of blends and neat materials. A neat IHIC-N film shows no obvious scattering peaks (Figure S13a), indicating low crystallinity, as intended by our choice of alkyl, rather than aryl, core substituents. The PTB7-Th:IHIC-N film (Figure 4a) shows a strong lamellar peak concentrated along the q_r axis ($q_r =$

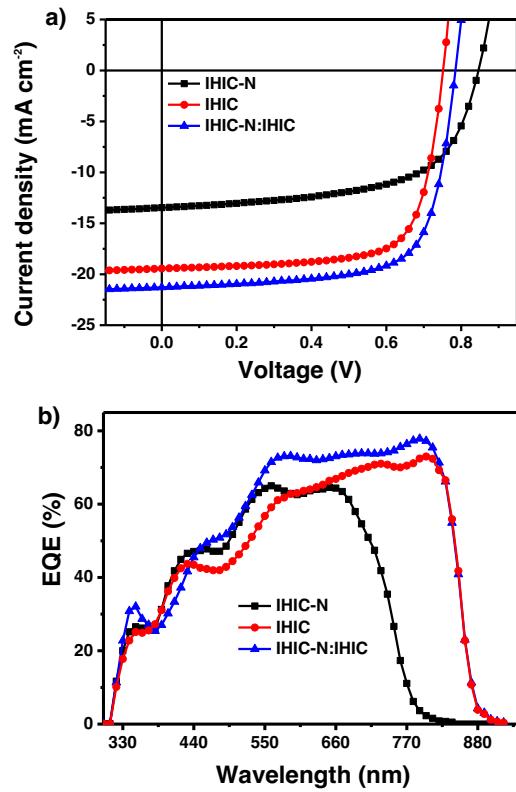


Figure 3. (a) J - V characteristics and (b) EQE spectra of the best PTB7-Th:NFA OSCs (1:1.5) under AM 1.5 G at 100 mW cm⁻². For the ternary cells the IHIC-N:IHIC ratio is 1:2.

0.284 \AA^{-1} , $d = 2.21 \text{ nm}$) and a peak attributed to π -stacking concentrated along the q_z axis ($q_z = 1.73 \text{ \AA}^{-1}$, $d = 0.362 \text{ nm}$). These features differ somewhat from those of neat PTB7-Th for which lamellar and π - π peaks are at $q = \text{ca. } 0.30$ and 1.60 \AA^{-1} , respectively;²⁷ indeed, while neat PTB7-Th shows a strong feature attributed to aggregation at ca. 710 nm in its UV-vis. absorption spectrum,²⁸ only a weak shoulder is seen in this region for PTB7-Th:IHIC-N, also suggesting

IHIC-N disrupts the polymer crystallinity. The observed π - π peak might be to the scattering of IHIC-N crystalline “face-on” domains, but cannot be reliably assigned as such since neat IHIC-N exhibits poor crystallinity and so its lattice constants are unknown. The PTB7-Th:IHIC film (Figure 4b) exhibits two lamellar peaks ($q_r = 0.278$ and 0.354 \AA^{-1}) and two π - π peaks ($q_z = 1.60$ and 1.80 \AA^{-1}), suggesting that both the PTB7-Th and IHIC domains are crystalline, both with preferred “face-on” orientation. The observation of polymer scattering is consistent with the observation of the PTB7-Th aggregate peak in the absorption spectrum of the blend.²⁸ The π -stacking distance assigned to IHIC (0.349 nm) is somewhat smaller than that seen for the PTB7-Th:IHIC-N blend (0.362 nm). Similar broadened π - π scattering features along the q_z axis are observed in the scattering pattern of the ternary PTB7-Th:IHIC-N:IHIC film,

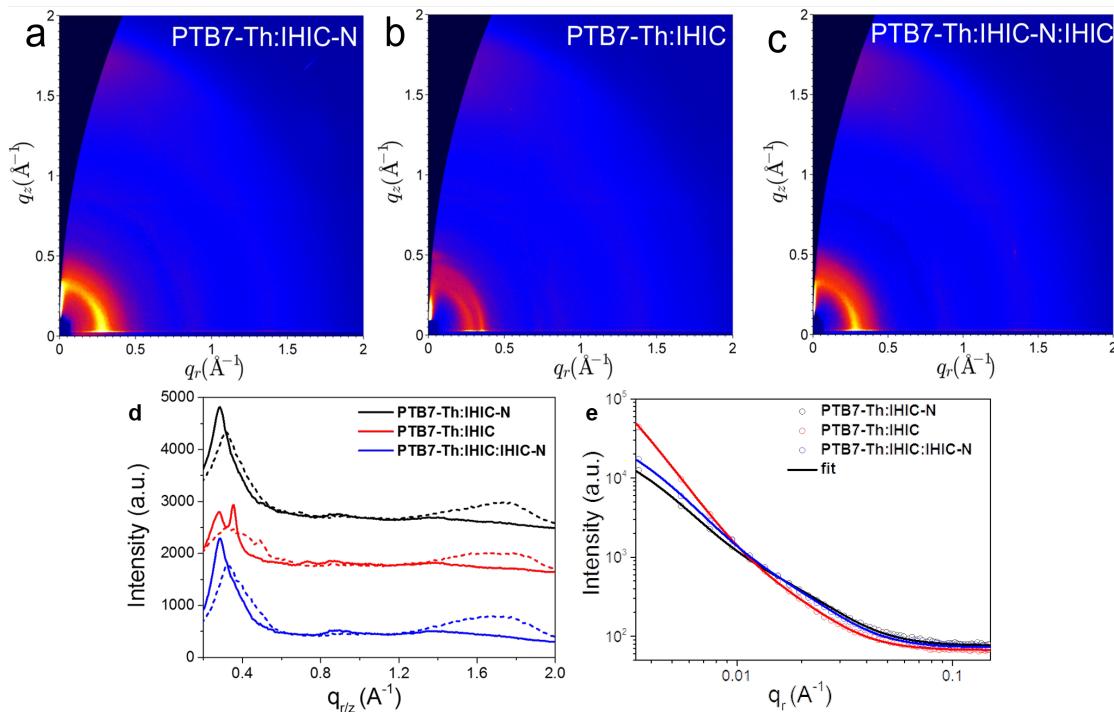


Figure 4. 2D GIWAXS patterns of (a) PTB7-Th:IHIC-N, (b) PTB7-Th:IHIC, (c) PTB7-Th:IHIC-N:IHIC (2:1:2), and (d) the corresponding GIWAXS intensity profiles along the in-plane (solid lines) and out-of-plane (dotted line) directions. (e) GISAXS intensity profiles and best fittings along the in-plane direction.

In summary, we have obtained a new NFA, IHIC-N. PTB7-Th:IHIC-N:IHIC OSCs based a PCE of 11.9%, due to balanced carrier mobilities, favorable domain sizes, and broad absorption, resulting in a panchromatic photovoltaic response from 300-900 nm. These results also highlight the ability of C-H functionalization to access compounds that are difficult to obtain by other routes, but which are desirable to fulfill specific requirements.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website.

Detailed experimental procedures and additional characterization data (solution UV-vis. absorption and fluorescence

suggesting “face-on” π - π stacking for both donor (PTB7-Th) and acceptor (IHIC and/or IHIC-N) materials, consistent with the more balanced carrier mobility observed in the ternary film. Figure 4e shows the GISAXS intensity profiles and best fittings along the in-plane direction. We adopted Debye–Anderson–Brumberger (DAB) and fractal-like network models to account for the scattering contribution from intermixed amorphous phases and acceptor domains respectively.²⁹ The acceptor domain sizes ($2R_g$) are estimated to be 21.4, 22.6 and 22 nm for PTB7-Th:IHIC-N, PTB7-Th:IHIC and PTB7-Th:IHIC-N:IHIC films, respectively, which would allow efficient exciton dissociation. The corresponding correlation lengths of the amorphous intermixed phases are fitted to be 26.9, 58.9, 27.5 nm, broadly consistent with TEM images (Figure S12), which show increasing domain size with increasing IHIC content.

spectra; electrochemical data; DFT molecular orbitals; SCLC, AFM, TEM, GIWAXS, GISAXS data; and NMR spectra) (PDF)

AUTHOR INFORMATION

Corresponding Author

*xwzhan@pku.edu.cn

*seth.marder@chemistry.gatech.edu

Author Contributions

#These authors contributed equally.

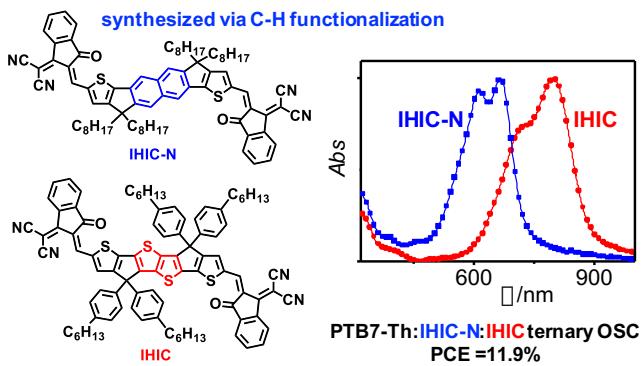
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C-H bond functionalization used to obtain key intermediate

Wider bandgap and lower electron affinity than IHIC

11.9% PCE in ternary devices