

**Predicting the glass transition temperature and viscosity of secondary
organic material using molecular composition**

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Abstract:

Secondary organic aerosols (SOA) account for a large fraction of submicron particles in the atmosphere. SOA can occur in amorphous solid or semi-solid phase states depending on chemical composition, relative humidity (RH), and temperature. The phase transition between amorphous solid and semi-solid states occurs at the glass transition temperature (T_g). We have recently developed a method to estimate T_g of pure compounds containing carbon, hydrogen, and oxygen atoms (CHO compounds) with molar mass less than 450 g mol⁻¹ based on their molar mass and atomic O:C ratio. In this study, we refine and extend this method for CH and CHO compounds with molar mass up to ~1100 g mol⁻¹ using the number of carbon, hydrogen, and oxygen atoms. We predict viscosity from the T_g -scaled Arrhenius plot of fragility (viscosity vs. T_g/T) as a function of the fragility parameter D . We compiled D values of organic compounds from literature, and found that D approaches a lower limit of ~10 (+/- 1.7) as the molar mass increases. We estimated viscosity of α -pinene and isoprene SOA as a function of RH by accounting for hygroscopic growth of SOA and applying the Gordon-Taylor mixing rule, reproducing previously published experimental measurements very well. Sensitivity studies were conducted to evaluate impacts of T_g , D , hygroscopicity parameter (κ), and the Gordon-Taylor constant on viscosity predictions. Viscosity of toluene SOA was predicted using the elemental composition obtained by high-resolution mass spectrometry (HRMS), resulting in a good agreement with the measured viscosity. We also estimated viscosity of biomass burning particles using the chemical composition measured by HRMS with two different ionization techniques: electrospray ionization (ESI) and atmospheric pressure photoionization (APPI). Due to differences in detected organic compounds and signal intensity, predicted viscosities at low RH based on ESI and APPI measurements differ by 2-5 orders of magnitude. Complementary

measurements of viscosity and chemical composition are desired to further constrain RH-dependent viscosity in future studies.

1. Introduction

Secondary organic aerosols (SOA) account for a large fraction of submicron particles in the atmosphere and they play an important role in climate, air quality and public health (Goldstein and Galbally, 2007; Jimenez et al., 2009). Traditionally, SOA particles were assumed to be liquid with dynamic viscosity η below 10^{-2} Pa s, but a number of recent studies have shown that they can also adopt amorphous semi-solid ($10^{-2} \leq \eta \leq 10^{-12}$ Pa s), or glassy solid ($\eta > 10^{-12}$ Pa s) states, depending on chemical composition and temperature (Zobrist et al., 2008; Koop et al., 2011; Huang et al., 2018; Reid et al., 2018). The phase state is also strongly affected by relative humidity, as water can act as a plasticizer to lower viscosity (Mikhailov et al., 2009). Ambient and laboratory-generated SOA particles have been observed to bounce off the smooth hard surface of an inertial impactor at low RH, implying a non-liquid state (Virtanen et al., 2010; Saukko et al., 2012; Bateman et al., 2015; Jain and Petrucci, 2015), whereas predominantly biogenic SOA particles in the Amazon basin did not bounce off the impactor surface at high RH, implying they are primarily liquid (Bateman et al., 2016). Upon dilution or heating, SOA particles were observed to evaporate unexpectedly slowly (Cappa and Wilson, 2011; Vaden et al., 2011), and recent modeling studies have evaluated the contributions of low diffusivity and volatility to slow evaporation rates (Roldin et al., 2014; Yli-Juuti et al., 2017). Measurements of viscosity of SOA bulk material derived from oxidation of α -pinene (Renbaum-Wolff et al., 2013; Zhang et al., 2015; Hosny et al., 2016), limonene (Hinks et al., 2016), isoprene (Song et al.,

2015), and toluene (Song et al., 2016a) have confirmed that SOA particles adopt a wide range of viscosities.

Viscosity can be directly converted to bulk diffusivity of organic molecules using the Stokes-Einstein equation (Einstein, 1905; Atkins, 1998; Seinfeld and Pandis, 2006; Schmelzer and Gutzow, 2011). This equation has been shown to work well for organic molecules diffusing through materials with η below $\sim 10^{-3}$ Pa s (Price et al., 2016; Chenyakin et al., 2017). Note that this relation is not accurate for predicting the bulk diffusivity of water and small molecules and it may also underestimate diffusivity of organic molecules in highly viscous matrix by a few orders of magnitudes (Champion et al., 2000; Shiraiwa et al., 2011; Power et al., 2013; Marshall et al., 2016; Bastelberger et al., 2017; Reid et al., 2018). The particle phase state, viscosity and bulk diffusivity have been shown to affect gas uptake and chemical transformation of organic compounds due to kinetic limitations of bulk diffusion (Shiraiwa et al., 2011; Abbatt et al., 2012; Kuwata and Martin, 2012; Zhou et al., 2013; Slade and Knopf, 2014; Arangio et al., 2015; Davies and Wilson, 2015; Wang et al., 2015; Berkemeier et al., 2016; Marshall et al., 2016; Liu et al., 2018; Pratap et al., 2018; Zhang et al., 2018), which may facilitate long-range transport of organic compounds embedded in viscous or glassy particles (Shrivastava et al., 2017b; Mu et al., 2018). Molecular motion can be hindered in a highly viscous matrix, slowing down photochemical reactions in particles (Lignell et al., 2014; Hinks et al., 2016). Water diffusion can be still fast even in an amorphous solid matrix under room temperature, but it can be hindered significantly under low temperatures (Mikhailov et al., 2009; Zobrist et al., 2011; Bones et al., 2012; Berkemeier et al., 2014; Price et al., 2014), affecting homogeneous vs. heterogeneous ice nucleation pathways (Murray et al., 2010; Wagner et al., 2012; Wang et al., 2012a; Wang et al., 2012b; Wilson et al., 2012; Baustian et al., 2013; Schill and Tolbert, 2013; Berkemeier et al.,

2014; Schill et al., 2014; Lienhard et al., 2015; Ignatius et al., 2016; Knopf et al., 2018). Despite the substantial implications of the SOA particle phase state, its effects on gas-particle interactions have not yet been considered explicitly in current climate and air quality models (Shrivastava et al., 2017a).

Partitioning of semi-volatile compounds into viscous particles may result in kinetically-limited growth in contrast to quasi-equilibrium growth (Perraud et al., 2012; Booth et al., 2014; Zaveri et al., 2014), which also affects the evolution of particle size distribution upon SOA growth (Shiraiwa et al., 2013; Zaveri et al., 2018). Note that the equilibration timescale of SOA partitioning is determined by bulk diffusivity or viscosity, but also affected by other factors such as volatility, accommodation coefficient, particle size and mass loadings (Shiraiwa and Seinfeld, 2012; Mai et al., 2015; Liu et al., 2016). Chamber experiments probing mixing timescales of SOA particles derived by oxidation of various precursors such as isoprene, terpene, and toluene have observed strong kinetic limitations at low RH, but not at moderate and high RH (Loza et al., 2013; Ye et al., 2016; Ye et al., 2018). Several studies have observed kinetic limitations of bulk diffusion of organic molecules including polycyclic aromatic hydrocarbons (Abramson et al., 2013; Zhou et al., 2013) and isoprene derived epoxydiols (Zhang et al., 2018) in SOA, while Gorkowski et al. (2017) did not observe significant diffusion limitations for glycerol and squalene in α -pinene SOA. Quasi-equilibrium versus kinetically-limited or non-equilibrium SOA growth remains an open issue and warrants further investigations.

Group contribution methods have been used to predict the viscosities of pure compounds when the functionality and molecular structure are known (Sastri and Rao, 1992; Rothfuss and Petters, 2017a). Song et al. (2016b) showed that estimations from group contribution approaches combined with either non-ideal or ideal mixing reproduced the RH-dependent trends particularly

well for the alcohol, di-, and tricarboxylic acid systems with viscosity of up to 10^4 Pa s. In contrast, model calculations overestimated the viscosity of more viscous compounds including mono-, di-, and trisaccharides by many orders of magnitude (Song et al., 2016b). A recent study compiled viscosity of organic compounds with atmospherically relevant functional groups, investigating the influence of the number and location of functional groups on viscosity (Rothfuss and Petters, 2017a). These studies provide important insights in estimating the viscosity of individual organic compounds.

Particle phase state can be characterized by a glass transition temperature (T_g), which is a characteristic temperature representing a non-equilibrium phase transition from a glassy solid state to a semi-solid state as the temperature increases (Koop et al., 2011). Recently, we have developed a parameterization to estimate T_g of pure organic compounds comprised of carbon, hydrogen, and oxygen (CHO compounds) with molar mass less than 450 g mol^{-1} based on their molar mass and atomic O:C ratio (Shiraiwa et al., 2017). It has been applied successfully in a global chemistry climate model to predict T_g and the phase state of atmospheric SOA, which indicated that SOA particles are mostly liquid or semi-solid in the planetary boundary layer, while they should be glassy in the middle and upper troposphere (Shiraiwa et al., 2017). A recent study provided a consistent result, suggesting that mixing timescales of organic molecules within SOA are often $< 1 \text{ h}$ in a global planetary boundary layer (Maclean et al., 2017).

It has been shown that SOA particles contain oligomeric compounds with molar masses higher than 450 g mol^{-1} (Gao et al., 2004; Tolocka et al., 2004; Nizkorodov et al., 2011; Nozière et al., 2015), which makes the previously developed parameterization incomplete. In this study, we extend the parameterization of T_g to higher molar mass compounds, and apply it to high-resolution mass spectrometry data for toluene SOA and biomass burning particles. The

Arrhenius approach and the Gordon-Taylor mixing rules were applied to estimate viscosity of SOA bulk materials to compare with the literature reported viscosity measurements. This method will be useful for estimations of viscosity of organic particles, for which high-resolution mass spectra are available. It can also be applied in global or regional models to evaluate impacts of the particle phase state on the role of SOA in climate and air quality.

2. Parameterization development

2.1 Glass transition temperature

Figure 1a shows the dependence of T_g on the molar mass (M) of organic compounds. Solid markers represent measured T_g of 258 CHO compounds (Koop et al., 2011; Dette et al., 2014; Rothfuss and Petters, 2017a), while open markers represent 654 CHO compounds in SOA (Shiraiwa et al., 2014). Markers are color-coded by atomic O:C ratio. Their melting points (T_m) were estimated by the Estimation Programs Interface (EPI) Suite software version 4.1 (US-EPA, 2012) and their T_g were estimated using the Boyer-Kauzmann rule: $T_g = g \cdot T_m$ with $g = 0.7$ (Koop et al., 2011; Shiraiwa et al., 2017). This rule can provide good estimates of T_g as has been validated in previous work (Koop et al., 2011) and also shown in Fig. A2(a). A subset of data shown in Figure 1 was originally published in Shiraiwa et al. (2017) for compounds with $M < 450 \text{ g mol}^{-1}$. This version of the figure has been updated to include a number of experimentally measured T_g values of larger compounds with M up to 1153 g mol^{-1} , including aliphatic compounds containing OH and/or COOH groups. Specifically, data for 76 aliphatic alcohols, 39 carbohydrates and their derivatives, 4 carboxylic acids, and 4 hydroxy acids, as compiled by Rothfuss and Petters (2017b), have been added to Figure 1. Eight of these compounds are carbohydrates with $M > 450 \text{ g mol}^{-1}$. These updates are critical for reliable parameterization of T_g

based on M . When M increases above $\sim 500 \text{ g mol}^{-1}$, the slope of T_g decreases, making it challenging to extrapolate the low- M data from the original Shiraiwa et al. (2017) study to higher M values. When M increases to $\sim 1000 \text{ g mol}^{-1}$, the corresponding T_g appears to level at around 420 K.

Such dependence on M has been described for polymers with the Fox-Flory equation: $T_g(M) = T_{g,\infty} - \frac{K_m}{M}$ (Fox Jr and Flory, 1950), where K_m is a constant and $T_{g,\infty}$ is the asymptotic value of T_g specific to the polymer. We conducted a literature search and found that most of the reported $T_{g,\infty}$ values fell below $\sim 500 \text{ K}$ (Fox Jr and Flory, 1950; Onder et al., 1972; Montserrat and Colomer, 1984; Polymer handbook, 1999; Papadopoulos et al., 2004; Matsushima et al., 2017). The Fox-Flory equation works very well for high molar mass compounds and is also generally applicable to smaller compounds (Koop et al., 2011), as supported by an approximately linear dependence of T_g on the inverse molar mass in Fig. A1(a). Figure 1b plots the values of T_g as a function of the atomic O:C ratio of organic molecules. Figures 1a and 1b clearly demonstrate that T_g depends primarily on the molar mass with a weak dependence on the atomic O:C ratio.

A parameterization for T_g calculation based on the molar mass and atomic O:C ratio was developed in our recent work, which is applicable to CH and CHO compounds with $M < 450 \text{ g mol}^{-1}$ (Shiraiwa et al., 2017):

$$T_g = A + BM + CM^2 + D (\text{O:C}) + E M (\text{O:C}) \quad (1)$$

where $A = -21.57 (\pm 13.47) [\text{K}]$, $B = 1.51 (\pm 0.14) [\text{K mol g}^{-1}]$, $C = -1.7 \times 10^{-3} (\pm 3.0 \times 10^{-4}) [\text{K mol}^2 \text{g}^{-2}]$, $D = 131.4 (\pm 16.01) [\text{K}]$ and $E = -0.25 (\pm 0.085) [\text{K mol g}^{-1}]$, respectively. These values were obtained by fitting the measured T_g of 179 CH and CHO compounds with $M < 450 \text{ g mol}^{-1}$ with multi-linear least squares analysis. Note that application of Eq. (1) may provide

unreasonable T_g values for compounds with $M > 500 \text{ g mol}^{-1}$ because it does not account for the strong curvature in the T_g vs. M dependence shown in Figure 1a.

In this study we have developed an improved parameterization to predict T_g of CH and CHO compounds using the number of carbon (n_C), hydrogen (n_H), and oxygen (n_O) that can also be applied to higher molar mass compounds. Motivated by a good correlation between T_g and volatility (Fig. 1a in Shiraiwa et al. (2017)), we use an equation with a similar formulation to the equation used to predict the saturation mass concentration or volatility (Donahue et al., 2011; Li et al., 2016):

$$T_g = (n_C^0 + \ln(n_C)) b_C + \ln(n_H) b_H + \ln(n_C) \ln(n_H) b_{CH} + \ln(n_O) b_O + \ln(n_C) \ln(n_O) b_{CO} \quad (2)$$

where n_C^0 is the reference carbon number, b_C , b_H and b_O denote the contribution of each atom to T_g , and b_{CH} and b_{CO} are coefficients that reflect contributions from carbon-hydrogen and carbon-oxygen bonds, respectively. These values were obtained by fitting the measured T_g of 42 CH compounds and 258 CHO compounds with multi-linear least squares analysis with 68% prediction and confidence intervals. The best-fit parameters are summarized in Table 1.

Note that the evaluation dataset used to derive Eq. (2) contains CH compounds with $M < 260 \text{ g mol}^{-1}$ (see Fig. A2b for comparison of measured and predicted T_g). Thus, the application of Eq. (2) to higher molar mass compounds may require further refinement of the method when measured T_g for higher molar mass CH compounds becomes available. Figure 1c shows that the T_g values predicted using Eq. (2) are in good agreement with the T_g values measured in experiments (see also Fig. A1(b)) or estimated by the Boyer-Kauzmann rule as indicated by the high correlation coefficient of 0.95. T_g of individual compounds can be predicted within $\pm 21 \text{ K}$ as indicated by the prediction band (dotted lines in Fig. 1c); however, this uncertainty may be

much smaller for multicomponent SOA mixtures under ideal mixing conditions as indicated in the confidence band (dashed lines, almost overlapping with the 1:1 line).

These results are noteworthy given that the parameterization (Eq. 2) does not consider either explicit molecular structures or functional groups. Previous studies have shown that T_g can be especially sensitive to the number of OH groups, which interact strongly through hydrogen bonding. For example, Nakanishi et al. (2011) found a direct relationship between T_g and the number of hydroxyl groups in a molecule for sugar alcohols; T_g increases as the number of OH groups increases. They reported that the correlation between T_g and the number of OH groups was much stronger than the correlation between T_g and the number of carbons in a molecule. Such a trend is implicitly included in Eq. (1) and (2), which contain the O:C ratio and number of oxygen atoms as parameters, respectively. Recently, Rothfuss and Petters (2017b) showed an approximately linear relationship between the number of OH groups and T_g for compounds with up to eight OH groups. Grayson et al. (2017) showed that addition of hydroxyl functional groups increases viscosity, a conclusion supported by both the experimental data and quantitative structure-property relationship model. The correlation between T_g and the number of carbon atoms is consistent with the free volume theory, in which molecular motion is restricted by the difference between the space required for a molecule to vibrate versus the space in which the molecule resides (i.e., the free volume) (White and Lipson, 2016). The correlation between T_g and the number of OH groups is more consistent with the topological constraint theory, where the primary influence is the three dimensional structure of the molecule as determined by molecular bonds and hydrogen-bonding networks (Nakanishi and Nozaki, 2011; van der Sman, 2013). Future experiments targeting more comprehensive T_g data, especially for higher molar mass compounds, would lead to further refinements of our T_g parameterizations.

Comparing Eq. (1) and (2), the two parameterizations give similar performance for compounds with $M < 450 \text{ g mol}^{-1}$ as shown in Fig. A2c. The statistical measures of correlation coefficient (R), mean bias (MB), and root mean square error (RMSE) are 0.93, -6.45 K , and 25.64 K , respectively, for the performance of Eq. (1), while for Eq. (2), they are 0.95, 3.15 K , and 21.11 K , respectively. It should be noted again that Eq. (1) cannot be used to predict T_g for compounds with $M > 450 \text{ g mol}^{-1}$. For example, T_g of stachyose ($M = 667 \text{ g mol}^{-1}$) predicted by Eq. (1) is 198 K , while that by Eq. (2) is 394 K , which agrees much better with the measured mean T_g of 396 K (Rothfuss and Petters, 2017a). Eq. (2) is more flexible than Eq. (1) and can be potentially expanded to include compounds containing hetero-atoms (e.g., nitrogen or sulfur), once substantial sets of experimental values of T_g for such compounds become available. Regarding the applications to air quality and climate models, Eq. (1) can be applied in the volatility basis set (VBS) (Donahue et al., 2006; Donahue et al., 2011) and the molecular corridor approach (Shiraiwa et al., 2014; Li et al., 2016) to predict the T_g of SOA particles (Shiraiwa et al., 2017), while the new parameterization may be suitable for coupling with the statistical oxidation model which characterizes the SOA evolution as a function of α and ω (Cappa and Wilson, 2012; Jathar et al., 2015).

These parameterizations (Eqs. 1, 2) calculate T_g based on the elemental composition of organic compounds. SOA particles contain a number of organic compounds as well as a variable amount of liquid water, which has low T_g (136 K) and can act as a plasticizer (Mikhailov et al., 2009; Koop et al., 2011). Under humid conditions, SOA particles take up water by hygroscopic growth in response to RH, lowering T_g and viscosity of SOA particles. Estimations of T_g for SOA-water mixtures were discussed by Shiraiwa et al. (2017), who applied the Gordon-Taylor equation validated for a wide range of mixtures of organics, polymers, and water (Roos, 1993;

Hancock and Zograf, 1994; Zobrist et al., 2008; Dette et al., 2014; Dette and Koop, 2015). Briefly, T_g of mixtures of SOA compounds under dry conditions ($T_{g,org}$) were calculated assuming the Gordon-Taylor constant (k_{GT}) of 1 (Dette et al., 2014): $T_{g,org} = \sum_i w_i T_{g,i}$, where w_i is the mass fraction of organic compound i , which can be derived using mass concentrations of SOA products. The Gordon-Taylor equation can also be applied to calculate T_g of organic-water mixtures considering the mass fraction of organics (w_{org}) in SOA particles (Koop et al., 2011):

$$T_g(w_{org}) = \frac{(1-w_{org})T_{g,w} + \frac{1}{k_{GT}}w_{org}T_{g,org}}{(1-w_{org}) + \frac{1}{k_{GT}}w_{org}} \quad (3)$$

w_{org} can be calculated using the mass concentrations of water (m_{H_2O}) and SOA (m_{SOA}) as $w_{org} = m_{SOA} / (m_{SOA} + m_{H_2O})$. m_{H_2O} can be estimated using the effective hygroscopicity parameter (κ) (Petters and Kreidenweis, 2007):

$$m_{H_2O} = \frac{\kappa \rho_w m_{SOA}}{\rho_{SOA} \left(\frac{1}{a_w} - 1 \right)} \quad (4)$$

The density of water (ρ_w) is 1 g cm^{-3} , the density of SOA particles (ρ_{SOA}) is assumed to be 1.2 g cm^{-3} (Kuwata et al., 2012), m_{SOA} is the total mass concentrations of SOA, and a_w is the water activity calculated as $a_w = RH/100$. Pajunoja et al. (2015) found that water uptake in subsaturated conditions is inhibited until RH is high enough for dissolution of water in SOA particles with relatively low O:C ratios. As oxidation of SOA increases, solubility of water increases and dissolution occurs at lower RH values. In both cases, the use of subsaturated hygroscopicity measurements was supported.

2.2 Viscosity

Temperature dependence of viscosity (η) can be predicted using the modified Vogel-Tammann-Fulcher (VTF) equation (Angell, 1991):

$$\eta = \eta_{\infty} e^{\frac{T_0 D}{T - T_0}} \quad (5)$$

where η_{∞} is viscosity at infinite temperature; T_0 is the Vogel temperature; T is the ambient temperature. The fragility parameter, D , characterizes how rapidly the dynamics of a material slow down as T approaches T_g , reflecting to what degree the temperature dependence of the viscosity deviates from Arrhenius behavior. When T is close to T_g ($T_g/T \approx 1$), smaller D values indicate that viscosity is sensitive to temperature change (fragile behavior); while larger D values indicate that viscosity is less sensitive to temperature change (strong or Arrhenius behavior).

Assuming $\eta_{\infty} = 10^{-5}$ Pa s (Angell, 1991):

$$\log \eta = -5 + 0.434 \frac{T_0 D}{T - T_0} \quad (6)$$

When $T = T_g$, $\eta = 10^{12}$ Pa s, which leads to (Angell, 1991; Angell, 2002):

$$T_0 = \frac{39.17 T_g}{D + 39.17} \quad (7)$$

As can be seen in Eq. (7), both T_g and D are required to calculate η from Eq. (6) at a given temperature.

Figure 2 shows the T_g -scaled Arrhenius plot of fragility (viscosity versus T_g/T) referred to as an Angell plot (Angell, 1995). D values of organic compounds are typically in the range of ~5–30 (Angell, 1997). To estimate D values that could be applied to SOA compounds, we compiled measured fragility values. Fragility was often measured in the form of the fragility steepness index (m), which represents the slope of the Arrhenius plot at the point where $T = T_g$ (Boehmer et al., 1993). Compounds with lower m exhibit higher D values, indicating stronger glass formers. The measured m of 95 organic compounds are included in the Supplement. m can be converted to D using the following equation (see the full derivation of this equation in Appendix A):

$$D = \frac{665.89}{m-17} \quad (8)$$

Figure 3 shows the measured D as a function of (a) molar mass and (b) atomic O:C ratio of organic molecules. The molar mass exerts a stronger effect on fragility, while there is little dependence of D on the O:C ratio. As molar mass increases, D approaches a lower limit of 10.3 (± 1.7), consistent with the value of 10 used in our recent study (Shiraiwa et al., 2017). To evaluate the impact of the variations of D on viscosity prediction, sensitivity calculations were conducted as described in Sect. 3.

Besides the VTF equation, another commonly used equation for describing the temperature dependence of viscosity is the Williams-Landel-Ferry (WLF) equation: $\log \frac{\eta(T)}{\eta(T_g)} = \frac{-C_1(T-T_g)}{C_2+(T-T_g)}$, where empirical parameters C_1 and C_2 are adopted as 17.44 and 51.6 K, respectively (Williams et al., 1955; Schill and Tolbert, 2013; Wang et al., 2015). The two equations are mathematically equivalent, both defined with respect to a reference temperature, and their parameters are related through $C_1 = \frac{DT_0}{2.303(T_g-T_0)}$ and $C_2 = T_g - T_0$. For the WLF equation, T_g is the reference temperature and there is a linear dependence assumed between temperature and free volume (O'Connell and McKenna, 1999; Huang and McKenna, 2001; Metatla and Soldera, 2007). For the VTF equation, the reference is the Vogel temperature (T_0)—a hypothetical temperature at which all non-vibrational motion ceases and viscosity becomes infinite and the theoretical foundation of the VTF equation includes both thermodynamic and kinetic considerations (O'Connell and McKenna, 1999; Huang and McKenna, 2001; Metatla and Soldera, 2007). Recently, Rothfuss and Petters (2017b) applied a similar approach to model viscosity for sucrose particles by applying the VTF and Gordon-Taylor approaches. The calculations of viscosity of multi-component SOA mixtures in this study are based mainly on the

VTF equation and the difference between calculated results from the two equations will be briefly discussed in the following section.

3. Comparison of predicted viscosity with measurements

3.1. SOA formed from α -pinene and isoprene

The purpose of this section is to demonstrate that viscosity of SOA material can be predicted over a broad range of RH values from four parameters: T_g of dry SOA ($T_{g,org}$), fragility (D), hygroscopicity (κ), and the Gordon-Taylor constant for mixing SOA and water (k_{GT}). Viscosity of α -pinene SOA has been measured or estimated as a function of RH by several groups using multiple experimental techniques as shown in Fig. 4(a) (Abramson et al., 2013; Renbaum-Wolff et al., 2013; Kidd et al., 2014; Pajunoja et al., 2014; Bateman et al., 2015; Zhang et al., 2015; Grayson et al., 2016). The wide range of experimentally measured viscosities reported for α -pinene SOA, particularly from 30-60% RH is most likely a consequence of the different experimental approaches, mass loadings and O:C ratios for each experiment. For instance, Grayson et al. (2016) used mass loadings of 121 to 14000 $\mu\text{g m}^{-3}$ and observed that viscosity decreased as mass loading increased. Higher mass loadings would lead to greater partitioning of semi-volatile and lower molar mass compounds into the particle phase, which would lead to the decrease of T_g and viscosity of the resulting SOA mixture, as very recently demonstrated experimentally by Jain et al. (2018). Grayson et al. (2016) concluded that their results should be considered a lower limit for viscosity of α -pinene SOA in the atmosphere. It should also be noted that the viscosity measurements from Renbaum-Wolff et al. (2013) were for the water-soluble portion of the SOA. These datasets suggest that viscosity of α -pinene SOA approaches very high values ($\sim 10^8$ Pa s) below 20-30% RH and decreases with an increase in

RH reaching a value of $\sim 10^6$ Pa s at 80% RH. As can be seen in Fig. 4(b), PAM-generated isoprene SOA is less viscous with $\eta < 10^6$ Pa s even under dry conditions, undergoing a phase transition from a semi-solid phase to a liquid phase at $\sim 55\%$ RH (Bateman et al., 2015; Song et al., 2015).

The solid lines with the shaded areas in Figure 4 are viscosity values predicted using $T_{g,org}$, D , κ , k_{GT} . $T_{g,org}$ values were adopted by Berkemeier et al. (2014) who estimated $T_{g,org}$ with the Boyer-Kauzmann rule using the melting point of representative SOA oxidation products. Note that Eq. (1) or (2) were not used to estimate $T_{g,org}$, which should be done in future studies by obtaining their elemental composition using high resolution mass spectrometry. For α -pinene, $T_{g,org}$ was assumed to be 278 K corresponding to an O:C ratio of 0.5 (Berkemeier et al., 2014), which is a typical O:C ratio of α -pinene SOA (Aiken et al., 2008; Chen et al., 2011; Putman et al., 2012).

The $T_{g,org}$ selected for isoprene SOA was 255 K, corresponding to the O:C ratio of 0.6. Although no measurements of the O:C ratio for the experimental isoprene SOA data were reported, Song et al. (2015) estimated O:C of 0.64 – 1.1 based on literature values. As O:C ratios are useful in estimating $T_{g,org}$, we encourage the measurement of the O:C ratio of SOA when conducting viscosity measurements. In contrast to α -pinene SOA, there are limited viscosity measurements for isoprene SOA. While the predicted viscosity is consistent with the experimental data, comparison of our model predictions to additional measurements is strongly recommended. Song et al. (2015) prepared isoprene SOA in a potential aerosol mass (PAM) reactor while the data by Bateman et al. (2015) were for isoprene SOA generated in a smog chamber. It has been suggested that under ambient conditions, the majority of isoprene-derived SOA can be derived through heterogeneous interactions with acidic sulfate particles forming

oligomers (Surratt et al., 2010; Lin et al., 2013; Gaston et al., 2014), which may increase viscosity compared to the model SOA generated in PAM or a chamber. Further studies are warranted to compare laboratory-generated and ambient isoprene SOA, and to investigate the effect of the acidic seed on the viscosity.

For both α -pinene and isoprene SOA, D was set to 10 based on the analysis presented in Fig. 3(a). κ was set to 0.1 based on field and laboratory measurements (Gunthe et al., 2009; Lambe et al., 2011b; Pajunoja et al., 2014; Petters et al., 2017) and k_{GT} was assumed to be 2.5 (Zobrist et al., 2008; Koop et al., 2011). Using these parameters, the predicted viscosities match well the magnitude and the RH-dependence of the measured viscosity of α -pinene and isoprene SOA. Figure 4 also shows predicted viscosities (dotted lines) using the WLF equation, which shows similar values as the VTF equation, but slightly underestimates the viscosity of α -pinene SOA at low RH and overestimates the viscosity of isoprene SOA at high RH.

Sensitivity studies were conducted to examine the effects of $T_{g,org}$, D , κ and k_{GT} , on the calculated viscosity. In these studies, $T_{g,org}$ of α -pinene and isoprene SOA were varied within 229 – 328 K and 255 – 316 K, respectively, representing $T_{g,org}$ of different oxidation states (Berkemeier et al., 2014). D was varied between 5 and 30, which is the range characteristic for organic compounds (see Fig. 3a). κ of 0.05 – 0.15 were used for α -pinene and isoprene SOA (Lambe et al., 2011b; Pajunoja et al., 2015). For the Gordon-Taylor constant, values of 2.5 ± 1.5 were considered (Zobrist et al., 2008; Koop et al., 2011; Dette et al., 2014; Dette and Koop, 2015).

The effect of varying each parameter on the calculated viscosity of α -pinene SOA is illustrated in Fig. 5. Variations of ± 50 K in $T_{g,org}$ result in 3-6 orders of magnitude difference in calculated values at dry conditions, indicating that $T_{g,org}$ is a critical parameter for viscosity

estimations. Decreasing D from 10 to 5 led to a decrease of calculated values by more than one order of magnitude. The calculated results were within the upper limit of measurements when increasing D from 10 to 20, and the predicted values were only slightly enhanced when further increasing D from 20 to 30. Calculated values with variations in κ from 0.05 to 0.15 and k_{GT} from 1.0 to 4.0 were all within the measured ranges.

For isoprene SOA, an increase of $T_{g,org}$ to 287 K, which represents a higher oxidation state (Berkemeier et al., 2014), led to calculated values to be several orders of magnitude higher than the upper limit of measurements (Fig. 6a). When $T_{g,org}$ reaches 316 K, isoprene SOA can occur as a solid for RH lower than ~40%. Compared to α -pinene SOA, a variation in D has a larger effect on the calculated viscosity (Fig. 6b). For a range of 5 – 30 for D , calculations with the D value of 10 agreed well with the measurements, while other D values resulted in calculated viscosity outside of the measured ranges. Figures 6c and 6d show that decreasing κ and k_{GT} below the reference values, the predictions overestimate the measured η by one or two orders of magnitude. The latter is most evident at RH > 60%, where the calculated values were higher than the upper limit of measurements. Modeling results with κ and k_{GT} increasing to 0.15 and 4.0, respectively, were within the lower limit of measurements.

The above comparison between the measured and predicted viscosity demonstrates that the method described in this study can reproduce reasonably well the measured RH-dependent viscosity of SOA formed from α -pinene and isoprene. The sensitivity calculations showed that $T_{g,org}$ contributed the most to the uncertainty in the viscosity estimates. Previous studies have shown that the experimental conditions such as particle mass concentrations (Grayson et al., 2016; Jain et al., 2018) and RH upon SOA formation (Kidd et al., 2014; Hinks et al., 2018) can

impact chemical composition of SOA and hence the phase state and viscosity. Further efforts to constrain the uncertainties are needed both in experiments and parameterizations.

3.2. SOA formed from toluene

In this and the following sections, we examine the feasibility of calculating the value of $T_{g,org}$ from mass spectrometry data on SOA. Hinks et al. (2017) measured the elemental composition of toluene SOA using nanospray desorption electrospray ionization high-resolution mass spectrometry (nano-DESI-HRMS) (Roach et al., 2010b, a). Toluene SOA were formed by OH photooxidation in an aerosol smog chamber at <2% RH (mass loading = $23 \mu\text{g m}^{-3}$) and 75% RH (mass loading = $8 \mu\text{g m}^{-3}$) to investigate the effect of RH on the chemical composition of toluene SOA formed under low- NO_x conditions. Measurements revealed a significant reduction in the fraction of oligomers present in toluene SOA generated under high RH conditions compared to SOA generated under low RH conditions (Hinks et al., 2017). The detected molar mass of individual oxidation products spanned a range of 102 - 570 g mol^{-1} at high RH, which increased up to 726 g mol^{-1} at low RH.

Figure 7(a) shows the interdependence of glass transition temperature, volatility, and molar mass of the detected toluene SOA compounds. Glass transition temperatures were calculated using Eq. (2). Saturation mass concentrations or volatilities of detected compounds were estimated from the elemental composition by using the parameterization of Li et al. (2016). The analysis is based on the molecular corridor approach—a two-dimensional framework of volatility and molar mass of SOA components constrained by boundary lines of low and high atomic O:C ratio, corresponding to n -alkanes ($\text{C}_n\text{H}_{2n+2}$, O:C = 0) and sugar alcohols ($\text{C}_n\text{H}_{2n+2}\text{O}_n$, O:C = 1), respectively (Shiraiwa et al., 2014; Li et al., 2016). The toluene SOA constituents are

well constrained by the molecular corridor and T_g are higher for compounds with higher molar mass and lower volatility.

Eq. (1) was used to calculate T_g for individual compounds with $M < 450 \text{ g mol}^{-1}$, while excluding compounds with molar mass higher than 450 g mol^{-1} . This approach was deemed reasonable as such high molar mass compounds account for $< 10\%$ of all toluene SOA products formed at low RH, and for $< 2\%$ formed at high RH. Eq. (2) was used to calculate T_g for all the detected compounds. T_g of dry toluene SOA ($T_{g,org}$) was then computed using the Gordon-Taylor approach with $k_{GT} = 1$ (Sect. 2.1). The relative mass concentrations of individual components were assumed to be proportional to their relative abundance in the nano-DESI-HRMS spectrum. This assumption has a number of caveats (Bateman et al., 2012; Nguyen et al., 2013), and as we will see below, it results in deviations between the predicted and measured viscosity. Table 2 summarizes the results of such calculations, showing that the $T_{g,org}$ by Eq. (1) – excluding high molar mass compounds – is about 10 K lower as compared to $T_{g,org}$ by Eq. (2). $T_{g,org}$ at low RH is predicted to be higher than $T_{g,org}$ at high RH, which results from a lower abundance of high molar mass compounds observed at high RH. This trend is consistent with Kidd et al. (2014), who showed that SOA material formed under dry conditions is more viscous than that formed under wet conditions.

Figure 7(b) shows the predicted viscosity of toluene SOA as a function of RH, as compared to the measured viscosity of toluene SOA formed in an oxidation flow reactor at 13% RH (Song et al., 2016a). Indirect viscosity measurements are also included in shaded boxes for toluene-derived SOA (Bateman et al., 2015; Li et al., 2015). Lines with shaded areas are calculated viscosities using $T_{g,org}$ as described above. κ was assumed to be 0.25 based on laboratory measurements (Lambe et al., 2011a; Hildebrandt Ruiz et al., 2015). To achieve good

fit, D was set to 13 and k_{GT} was assumed to be 3.0 (Dette et al., 2014). Estimations with Eq. (1) match the measured viscosity values very well over the entire RH range. Predictions with Eq. (2) overestimated the measurements by one or two orders of magnitude at moderate RH between 30% and 50%, while they agreed with the measurements derived at $RH \geq 60\%$ and at the dry conditions.

There are several possible reasons for the difference between the measurements and predictions. First, the relative abundance of high molar mass compounds observed in HRMS measurements may be overestimated, as high molar mass compounds tend to have higher (yet generally unknown) ionization efficiencies compared to lower molar mass compounds. Second, the nano-DESI-HRMS analysis of toluene SOA was limited to m/z range of 100 -1000 (Hinks et al., 2017). It is possible that some SOA products with lower molar mass were present in particles but not detected, which would lead to an overestimation of T_g . Third, the chemical composition of toluene SOA are likely different between Hinks et al. (2017) and Song et al. (2016a) because of the differences in the experimental conditions. Specifically, toluene SOA was formed in a Teflon chamber in Hinks et al., while Song et al. used an oxidation flow reactor to generate toluene SOA. The O:C ratios are 0.71 at low RH and 0.63 at high RH based on nano-DESI-HRMS measurements in Hinks et al. (2017), while the O:C ratio was 1.06 in Song et al. (2016a) based on the aerosol mass spectrometry (AMS) measurements.

In addition, different mass loadings may have affected viscosity. Song et al. (2016) measured viscosity at two different mass loadings (60-100 and 600-1000 $\mu\text{g m}^{-3}$) and compared their results to toluene SOA data in Bateman et al. (2015) (30-50 $\mu\text{g m}^{-3}$) and Li et al. (2015) (44-125 $\mu\text{g m}^{-3}$), observing little impact of mass loadings on viscosity. We carried out a sensitivity study of mass loadings on viscosity using a set of compounds detected by HRMS. The

saturation mass concentration was predicted for each component using the molecular corridor approach (Li et al., 2016). Assuming that the mass signal intensity is proportional to the total mass concentration of the compound in the mixture, and applying the absorptive partitioning theory (Pankow, 1994), particle-phase concentrations of each compound were predicted to estimate T_g at different organic aerosol mass loading values (1-1000 $\mu\text{g m}^{-3}$). The glass transition temperature of the SOA mixture decreases as mass loading increases. Viscosity decreases up to two orders of magnitude at low RH, while at high RH they have little difference as shown in Fig. A3. Simultaneous measurements of viscosity and chemical composition with different mass loadings should be performed in future studies.

3.3 Biomass Burning Particles

To further explore the applicability of our viscosity prediction method using elemental composition as measured by HRMS, we performed similar calculations for biomass burning organic particles emitted from test facility burns of subalpine fir and lodgepole pine trees, conducted as a part of the FIREX 2016 campaign (Selimovic et al., 2017). These samples were analyzed by HRMS using two different ionization sources: electrospray ionization (ESI) and atmospheric pressure photoionization (APPI). Mass spectra shown in Fig. 8(a) and (b) indicate that a substantial number of compounds were detected by both methods (109 and 170 compounds for subalpine fir and lodgepole pine, respectively). However, pronounced differences are also observed between the ESI and APPI spectra both in terms of the identity and signal intensities of the detected compounds.

Glass transition temperatures for the assigned CH and CHO compounds were computed using Eq. (2). Nitrogen and sulfur containing compounds (CHON and CHOS) are not yet

covered by Eq. (2) and were therefore excluded from the analysis. CHON and CHOS compounds comprised less than 10% of the detected ion intensity and <15% of the assigned compounds. Note that we do not intend to provide accurate estimates of viscosity of ambient biomass burning particles (as inorganic components are also not included in this analysis), but we investigate how the use of different ionization methods would lead to variations in our viscosity predictions. T_g of organic mixtures ($T_{g,org}$) were then calculated using the Gordon-Taylor approach with $k_{GT} = 1$, assuming that the relative concentration of each compound is proportional to its MS signal intensity. The calculated $T_{g,org}$ values for the mixtures are specified in the legend of Figure 9. For both types of mixtures, the calculated $T_{g,org}$ for the APPI MS data is lower than the value calculated based on the ESI MS data with a difference of 32 K for subalpine fir and 11 K for the lodgepole pine. Figure 9 shows the predicted viscosity as a function of RH, assuming $D = 10$, $\kappa = 0.10$ and $k_{GT} = 2.5$. The difference in $T_{g,org}$ derived from ESI and APPI results in a variation of predicted viscosity at low RH by up to five and two orders of magnitude for subalpine fir and lodgepole pine, respectively.

The difference in the calculated $T_{g,org}$ values is attributed to the chemical profile of the species detected using different ionization techniques as shown in mass spectra in Fig. 8(a) and (b). Van Krevelen diagrams in Fig. 8(c) and (d) illustrate these compositional differences between chemical species detected by ESI and APPI. ESI is more efficient at detection of polar compounds (Kiontke et al., 2016), which typically have higher O:C ratios and therefore would result in higher predicted values of glass transition temperature (Koop et al., 2011; Saukko et al., 2012). APPI enables the detection of nonpolar compounds with lower O:C ratios, in particular polycyclic aromatic hydrocarbons (PAHs), that have low ionization efficiencies when analyzed by ESI MS (Raffaelli and Saba, 2003; Itoh et al., 2006). Due to the complementary nature of

these ionization methods, it is most likely that the actual glass transition temperature and viscosity of each type of organic components in biomass burning aerosols are somewhere in between the values inferred from ESI and APPI data sets: ESI MS may be viewed as providing the upper limit of viscosity, while APPI MS gives the lower limit. Our results indicate that the use of complementary ionization techniques may help evaluate the associated uncertainty for the prediction of viscosity values based on the elemental composition as measured by HRMS.

4 Conclusions

We have developed a parameterization for calculation of the glass transition temperature of individual SOA compounds with molar mass up to $\sim 1100 \text{ g mol}^{-1}$ using the number of carbon, oxygen, and hydrogen atoms. Viscosity of SOA was estimated using the T_g -scaled Arrhenius plot of viscosity versus T/T_g and the Gordon-Taylor approach to account for mixtures of SOA and water. The fragility parameter D was compiled for organic compounds and we found that D approaches a lower limit of ~ 10 (± 1.7) as the molar mass increases. The resulting viscosity estimations agree well with measured viscosity of α -pinene and isoprene SOA, validating our method. Using HRMS data, glass transition temperatures of individual components and viscosity of toluene SOA were predicted, also resulting in a good agreement with measurements. However, we note that the predicted viscosities were higher than the measured values suggesting that additional considerations may need to be taken into account. For example, the ionization efficiency of both low and high molar mass compounds may have a pronounced effect on the relative abundance of different classes of compounds in HRMS data. The viscosity prediction method was also applied to biomass burning particles, whose elemental composition was measured using HRMS with two different ionization techniques. Substantial differences in

viscosity estimations were obtained using ESI and APPI mass spectra because these two ionization methods probe different subsets of compounds.

Figure 10 summarizes the predicted range of viscosity of α -pinene SOA, isoprene SOA (generated by PAM), toluene SOA, and biomass burning particles. Isoprene SOA has lower viscosity, reflecting lower glass transition temperature due to relatively low molar mass of isoprene oxidation products. α -pinene and toluene SOA have much higher viscosity with a different shape of the RH dependence due to differences in glass transition temperatures and hygroscopicity. Biomass burning particles have moderate viscosity between the two extreme cases. Currently, both predictions and measurements are subject to large uncertainties and variations. Complementary measurements of viscosity and chemical composition employing different ionization techniques are desired to further constrain RH-dependent viscosity in future studies. Current T_g parameterizations do not consider functionality or molecular structure explicitly and further measurements of T_g and viscosity of SOA would allow us to refine the method presented in this study. Nevertheless, current results offer a promising starting point and such simple parameterizations are practical for predicting viscosity of particles as measured by HRMS. The developed viscosity prediction method should also be useful in recent efforts of simulating the distribution of SOA phase state and related properties in regional or global air quality models (e.g., Maclean et al., 2017; Shiraiwa et al., 2017).

Appendix A: Conversion of fragility steepness index (m) to fragility (D)

Fragility steepness index (m) is defined as:

$$m = \lim_{T \rightarrow T_g} \frac{d \log \eta}{d(T_g/T)} \quad (\text{A1})$$

Combining Eq. (A1) with Eq. (6) gives:

$$m = \lim_{T \rightarrow T_g} \frac{d}{d(T_g/T)} (-5 + 0.434 \frac{T_0 D}{T - T_0}) \quad (\text{A2})$$

Considering that $\eta = 10^{12}$ Pa s at $T = T_g$ (Angell, 1991), and by defining $\Delta x = 1 - T_g/T$, and a combination with Eq. (7) leads to:

$$\begin{aligned} m &= \lim_{\Delta x \rightarrow 0} \frac{1}{\Delta x} \left(12 - (-5 + 0.434 \frac{\frac{39.17 T_g}{D + 39.17} D}{\frac{T_g}{1 - \Delta x} - \frac{39.17 T_g}{D + 39.17}}) \right) \\ &= \lim_{\Delta x \rightarrow 0} \frac{1}{\Delta x} \left(17 - 0.434 \frac{39.17 T_g D (1 - \Delta x)}{D T_g + 39.17 T_g \Delta x} \right) \\ &= \lim_{\Delta x \rightarrow 0} \frac{(665.89 + 17D)}{(D + 39.17 \Delta x)} \\ &= \frac{665.89 + 17D}{D} \quad (\text{A3}) \end{aligned}$$

Note that Eq. (A3) is derived assuming the high temperature limit of viscosity η_∞ is equal to 10^{-5} Pa s (Angell, 1991) in the VTF equation (Eq. 5). Similar equations for the relation between m and D were given by previous studies using different η_∞ and units (Angell et al., 1994; Angell, 2002; Bones et al., 2012) and applying those gave very similar results in our study.

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Table 1. Composition classes and then n_C^0 and b values (K) for glass transition temperature parameterizations obtained by least-squares optimization using the measurements compiled in Koop et al., (2011), Dette et al., (2014) and Rothfuss and Petters (2017b).

Classes	n_C^0	b_C	b_H	b_{CH}	b_O	b_{CO}
CH	1.96 (± 1.81)	61.99 (± 53.65)	-113.33 (± 44.47)	28.74 (± 20.86)		
CHO	12.13 (± 2.66)	10.95 (± 13.60)	-41.82 (± 14.78)	21.61 (± 5.30)	118.96 (± 9.72)	-24.38 (± 4.21)

Table 2. Glass transition temperatures calculated using Eq. (1) and (2) for toluene SOA produced at low relative humidity (< 2%) and high relative humidity (75%) conditions.

$T_{g,org}$ (K)	low RH	high RH
Equation (1)*	299	295
Equation (2)	313	303

* Compounds with $M > 450$ g mol⁻¹ were excluded from the analysis.

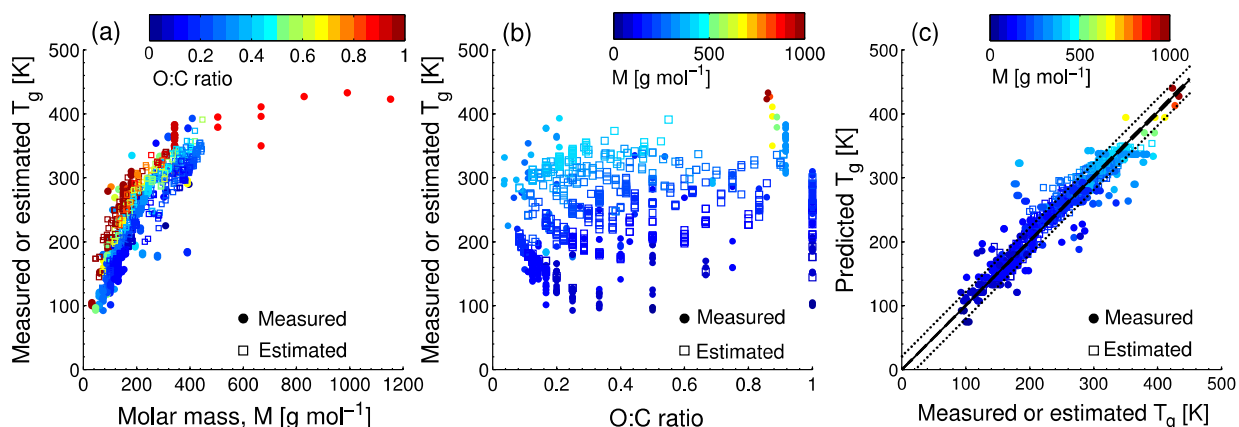


Figure 1. Characteristic relationships between molecular properties and the glass transition temperature (T_g) of organic compounds. (a) T_g of organic compounds as measured (circles) and estimated with the Boyer-Kauzmann rule (squares) plotted against molar mass. The markers are color-coded by atomic O:C ratio. (b) Measured (circles) and estimated (squares) T_g of organic compounds plotted against O:C ratio. The markers are color-coded by molar mass. (c) Predicted T_g for CHO compounds using a parameterization (Eq. 2) developed in this study compared to measured (circles) and estimated T_g by the Boyer-Kauzmann rule (squares). The solid line shows 1:1 line and the dashed and dotted lines show 68% confidence and prediction bands, respectively.

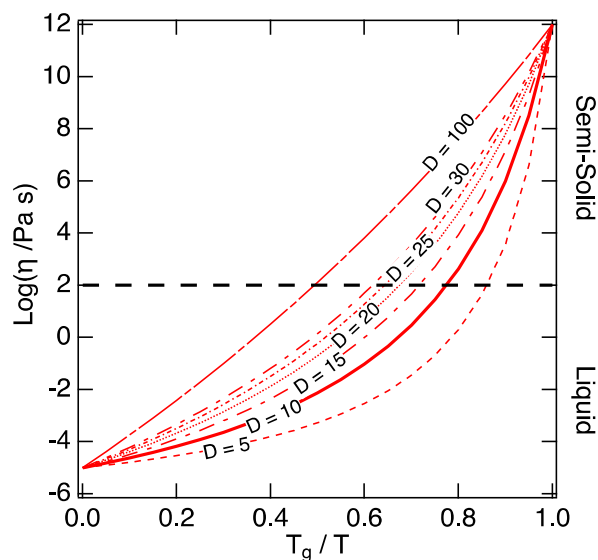
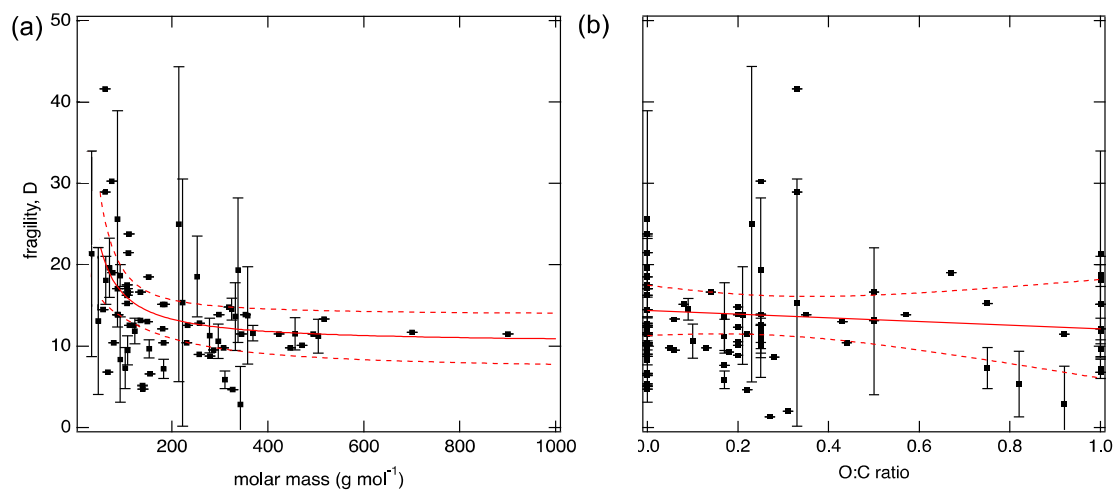
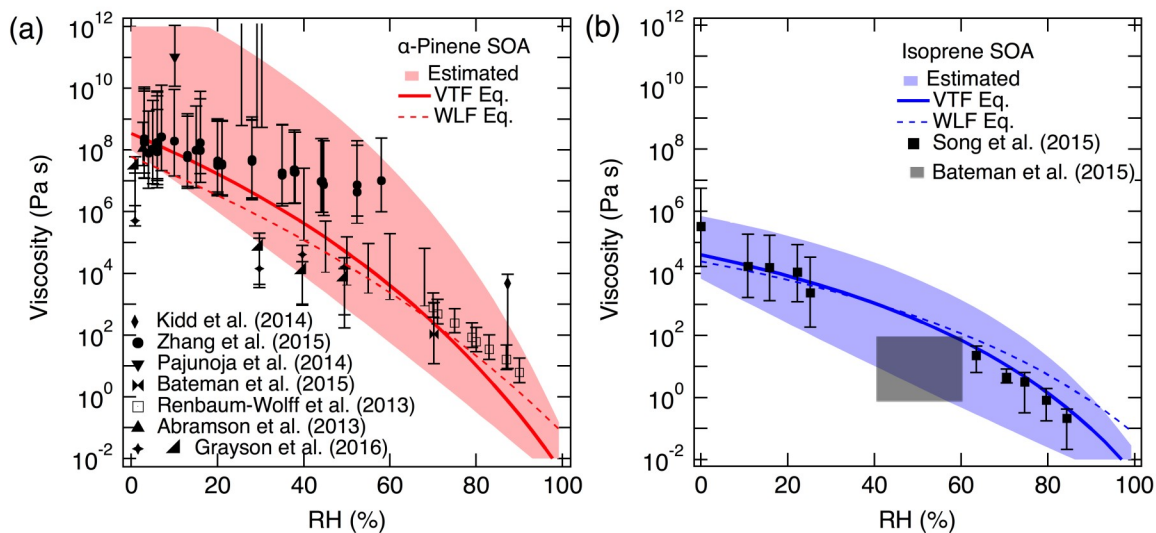


Figure 2. The Angell plot of viscosity (η) vs. T_g/T . The lines represent different fragility parameter (D) values in the range of 5 - 100, with $D = 10$ (the solid line) used as a base case for this study. A large fragility parameter value is associated with a strong glass former, while fragile materials are associated with lower values. The black dashed line at viscosity of 10^{-2} Pa s indicates the approximate threshold between liquid and semi-solid states.



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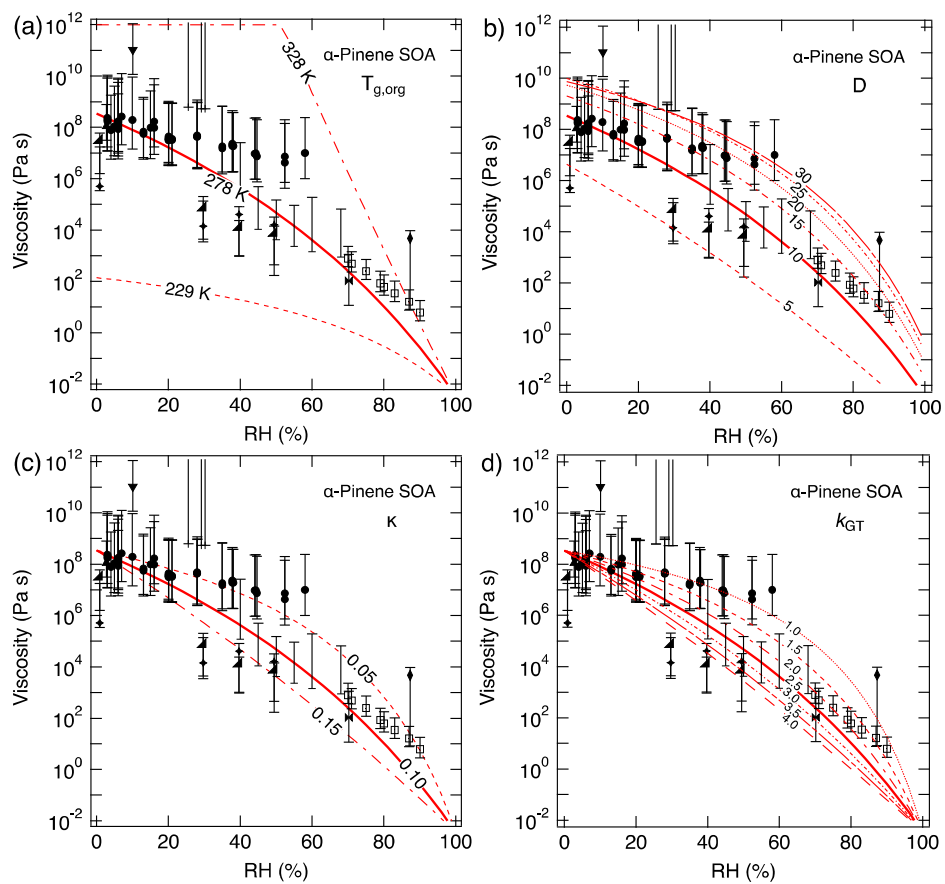


Figure 5. S varying: (a) glass transition temperature of dry SOA ($T_{g,org}$), (b) fragility (D), (c) hygroscopicity (κ), and (d) Gordon-Taylor constant (k_{GT}).

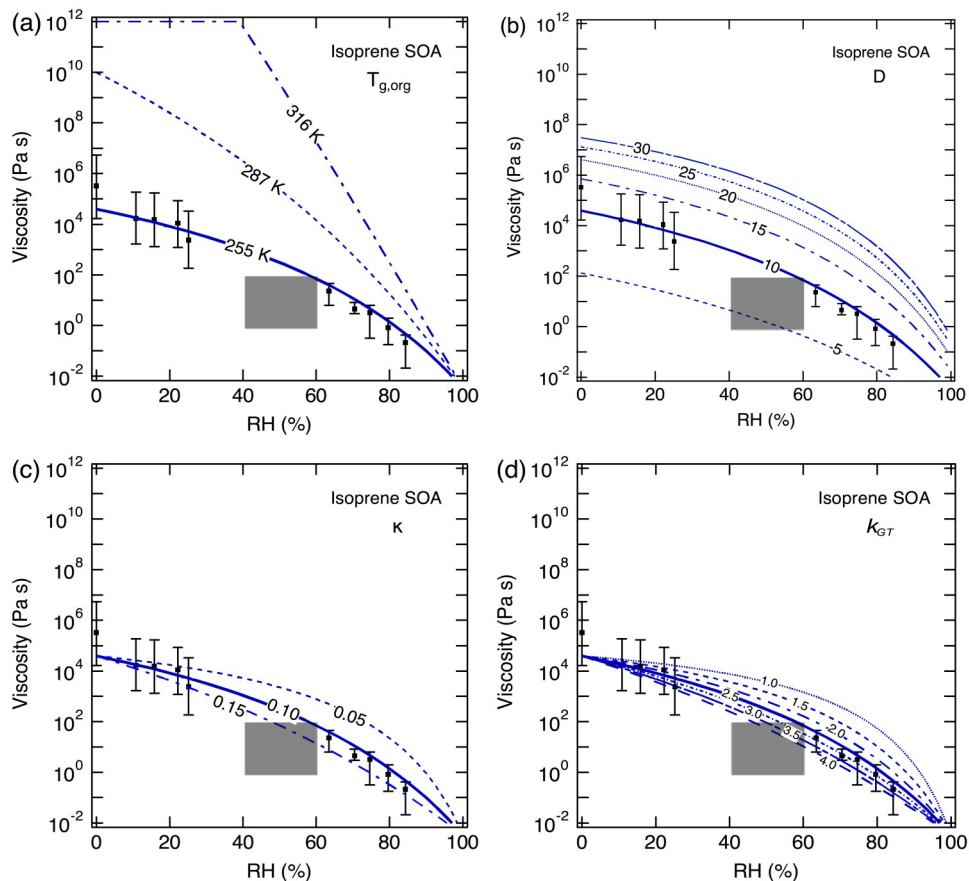


Figure 6. Sensitivity calculations for viscosity of isoprene SOA at 295 K as a function of RH by varying: (a) glass transition temperature of dry SOA ($T_{g,org}$), (b) fragility (D), (c) hygroscopicity (κ), and (d) Gordon-Taylor constant (k_{GT}). Data points are measured viscosity by Song et al. (2015) and the gray box represents estimated viscosity based on bounce measurements of Bateman et al. (2015).

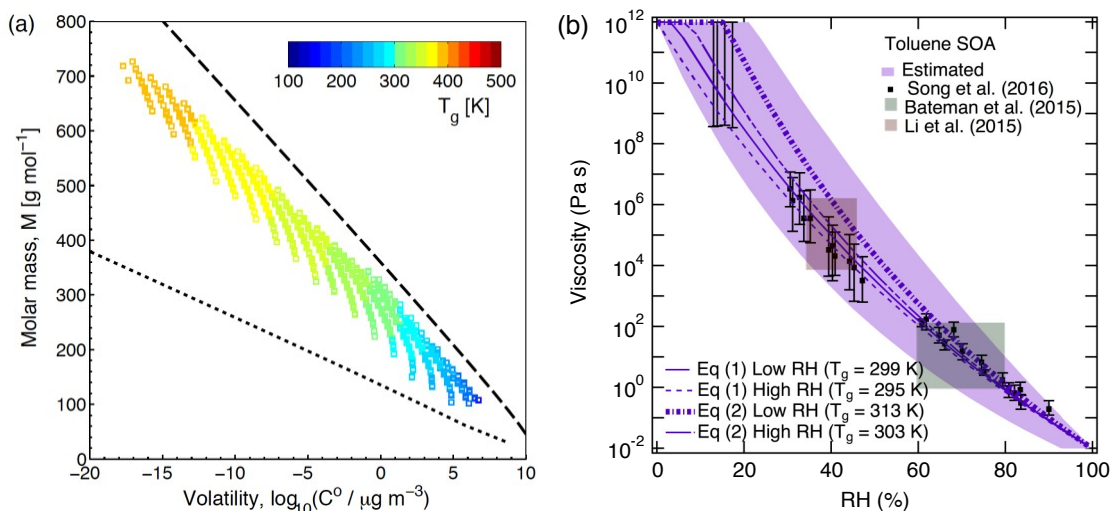


Figure 7. (a) Molecular corridor of molar mass plotted against volatility of toluene SOA formed under dry conditions (Hinks et al., 2017) color-coded by glass transition temperature (T_g) estimated using Eq. (2). The upper dashed line indicates the low O:C bound of the molecular corridor (linear alkanes C_nH_{2n+2} with O:C = 0), and the lower dotted line indicates the high O:C bound (sugar alcohols $C_nH_{2n+2}O_n$ with O:C = 1). (b) Comparison of measured (markers) and modeled (lines) viscosity of toluene SOA at 295 K as a function of RH. Viscosities were calculated using fragility (D) of 13, the hygroscopicity (κ) of 0.25 and the Gordon-Taylor constant (k_{GT}) of 3.0 with different glass transition temperatures of dry SOA ($T_{g,org}$) as estimated using Eq. (1) or (2) under low and high RH conditions. The shaded regions were calculated by varying those parameters: $T_{g,org} = 313$ K (295 K), $\kappa = 0.20$ (0.25), $D = 13$ (10), $k_{GT} = 2.5$ (3.5) for the upper (lower) limit. Mass loadings were $23 \mu\text{g m}^{-3}$ for Low RH and $8 \mu\text{g m}^{-3}$ for High RH (Hinks et al., 2017).

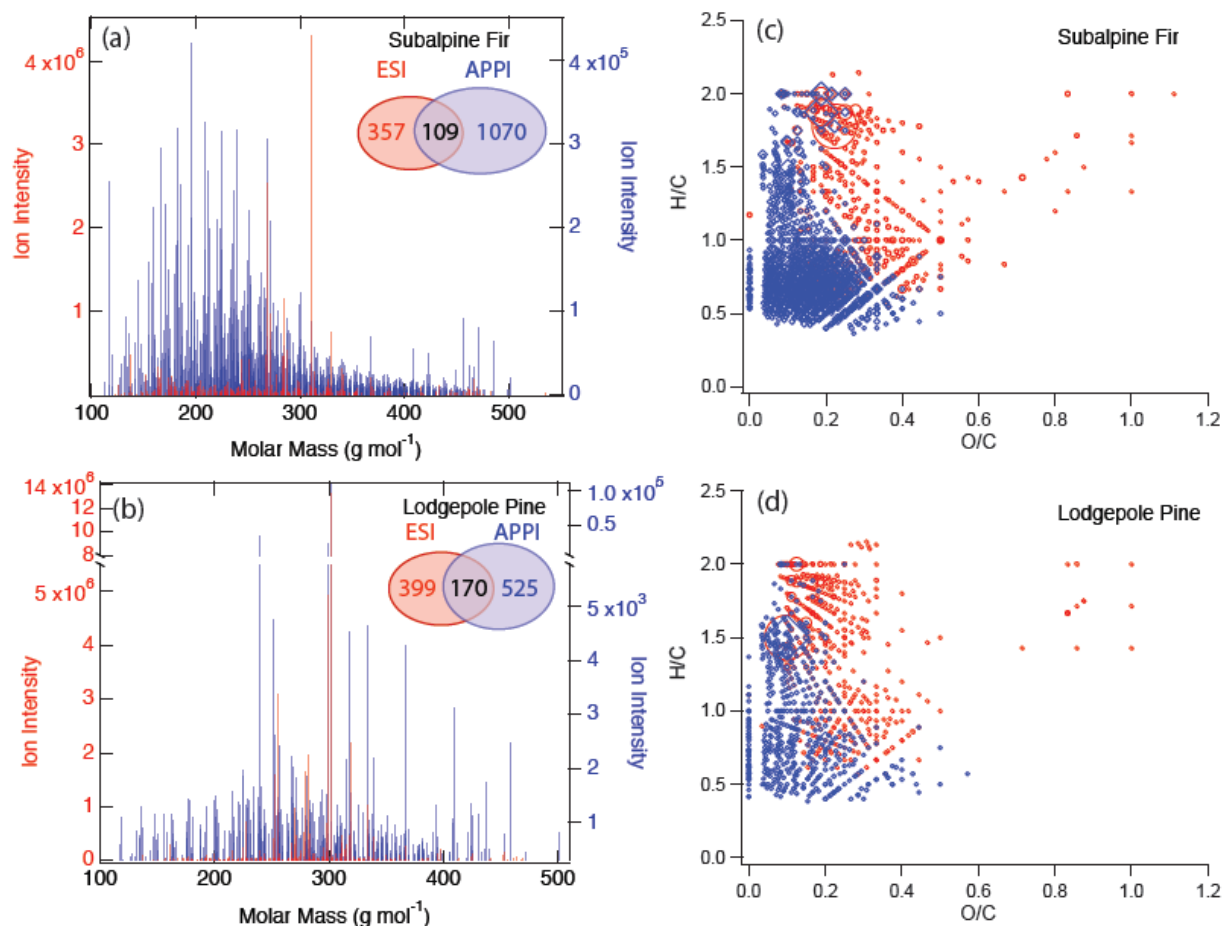


Figure 8. Mass spectra of biomass burning organic particles collected from test burns of (a) subalpine fir and (b) lodgepole pine as measured by high resolution mass spectrometry with two ionization techniques: electron spray ionization (ESI, red) and atmospheric pressure photoionization (APPI; blue). Numbers of elemental formulas identified by ESI (red), APPI (blue) and both modes (black) are also specified. Van Krevelen plots of the compounds identified by ESI (red) and APPI (blue) mode in BBOA from burning of (c) subalpine fir and (d) lodgepole pine.

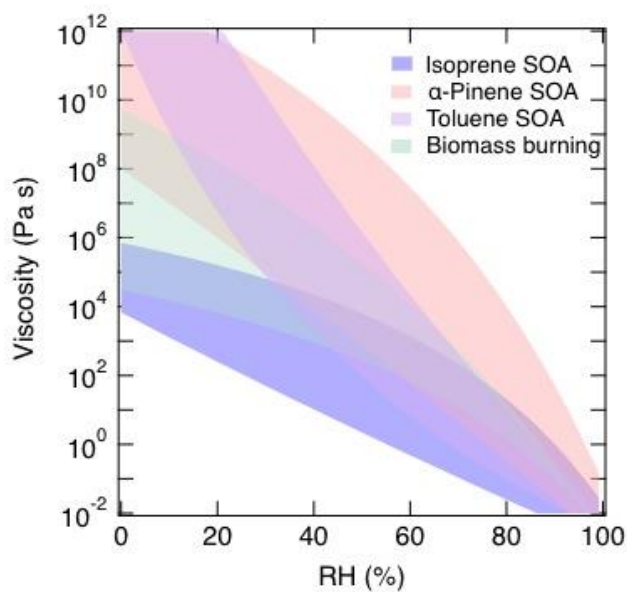
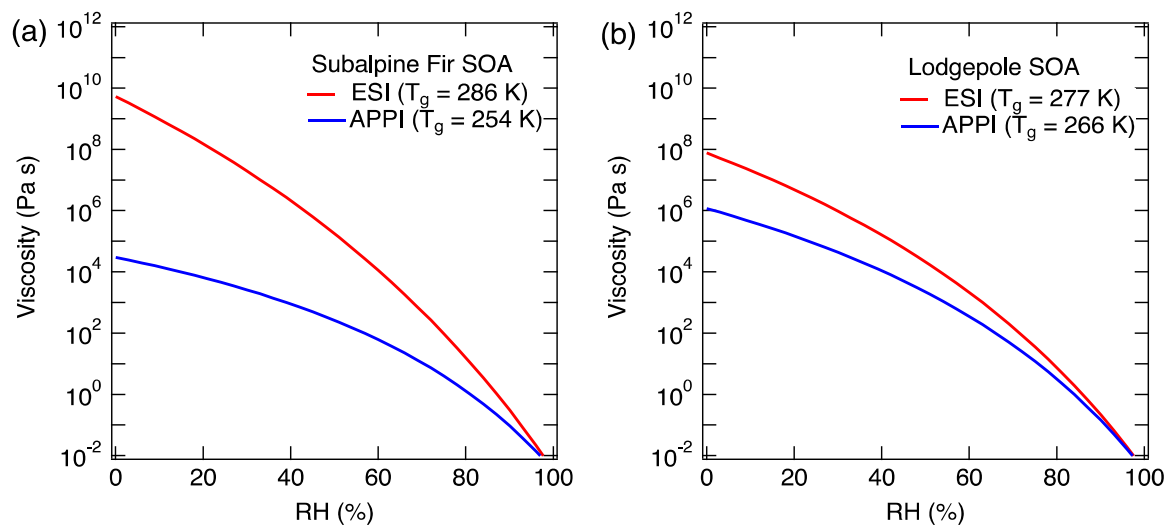


Figure 10. Summary of predicted range of viscosity of α -pinene SOA (red), isoprene SOA (blue), toluene SOA (purple), and biomass burning particles (green).

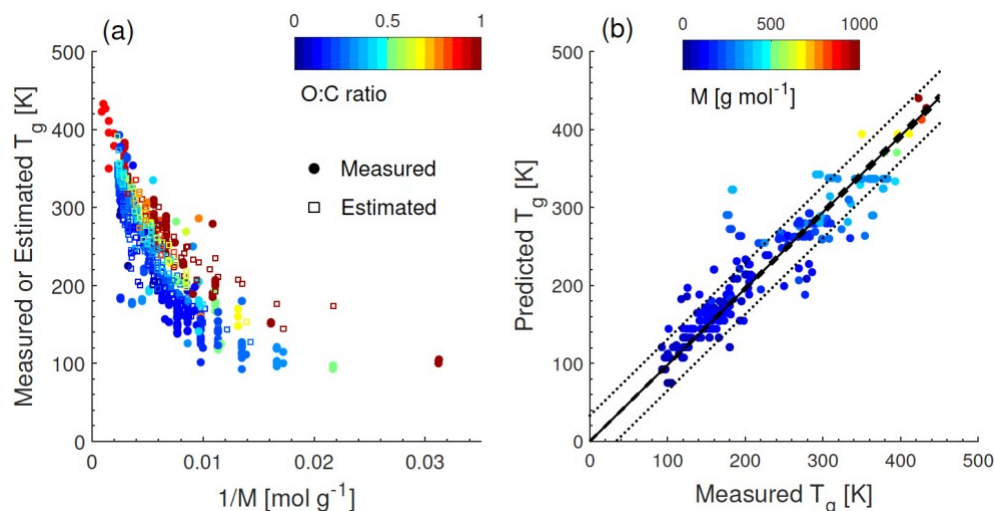


Figure A1. (a) T_g of organic compounds as measured (circles) and estimated with the Boyer-Kauzmann rule (squares) plotted against the inverse molar mass. The markers are color-coded by atomic O:C ratio. (b) Predicted T_g for CHO compounds using a parameterization (Eq. 2) developed in this study compared to measured T_g (circles). The solid line shows 1:1 line and the dashed and dotted lines show 68% confidence and prediction bands, respectively.

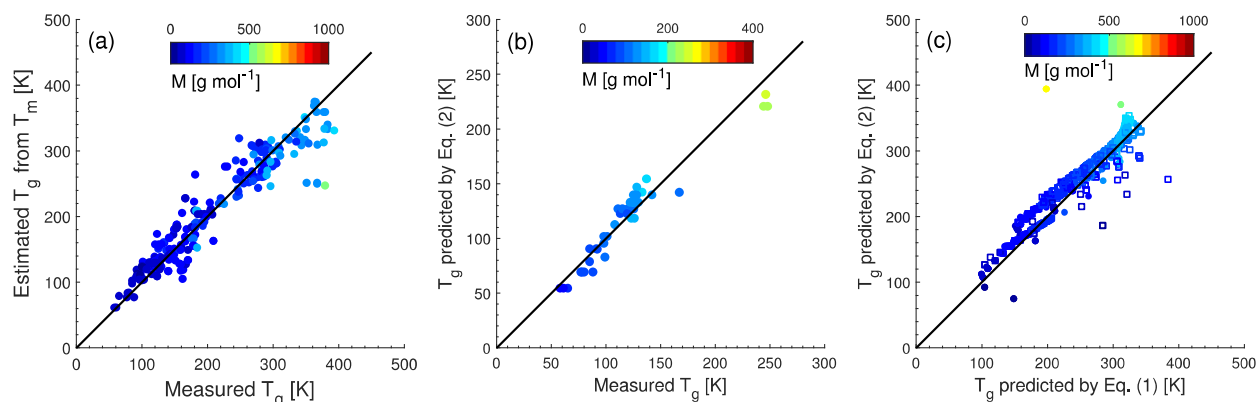


Figure A2. (a) Comparison of measured and estimated T_g by the Boyer-Kauzmann rule for 251 organic compounds (Koop et al., 2011; Dette et al., 2014; Rothfuss and Petters 2017b) with their measured T_m available. The markers are color-coded by molar mass. (b, c) Predicted T_g using Eq. (2) compared with (b) measured T_g for CH compounds and (c) predicted T_g using Eq. (1) for CHO compounds. The solid line shows 1:1 line. Solid circle markers represent organic compounds as compiled in Koop et al. (2011) and open square marker represent SOA oxidation products in Shiraiwa et al. (2014) in panel (c).

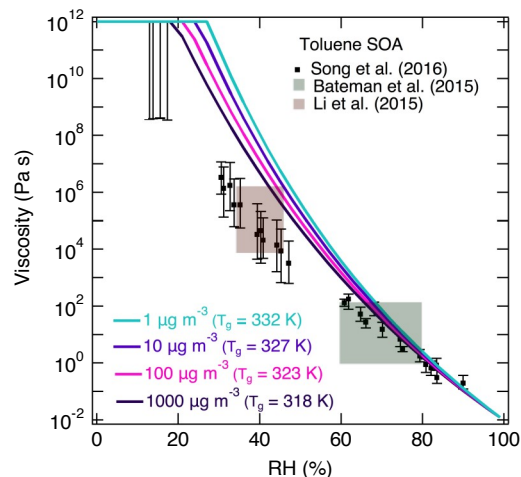


Figure A3. Effect of mass loading on predicted viscosity for toluene SOA. Solid lines represent the predicted viscosity with Eq. (2) using chemical composition of toluene SOA formed at low RH. Viscosity was predicted with different mass loadings ranging from 1-1000 $\mu\text{g m}^{-3}$. Markers and shaded boxes represent experimentally measured viscosity values. Song et al. (2016a) mass loadings were 60-100 and 600-1000 $\mu\text{g m}^{-3}$. Bateman et al., (2015) and Li et al., (2015) mass loadings were 30-50 $\mu\text{g m}^{-3}$ and 44-125 $\mu\text{g m}^{-3}$, respectively.