

1      **Delamination of a rigid punch from an elastic substrate under normal and shear forces**

2            XiaoHao Sun<sup>1,2</sup>, Luxia Yu<sup>2</sup>, Mark Rentschler<sup>2</sup>, HengAn Wu<sup>1\*</sup>, Rong Long<sup>2\*</sup>

3       <sup>1</sup>CAS Key Laboratory of Mechanical Behavior and Design of Materials, Department of  
4       Modern Mechanics, CAS Center for Excellence in Complex System Mechanics, University of  
5       Science and Technology of China, Hefei, Anhui 230027, China.

6       <sup>2</sup>Department of Mechanical Engineering, University of Colorado Boulder, Boulder, CO 80309,  
7       USA.

8       \*Correspondence should be addressed to R. Long ([rong.long@colorado.edu](mailto:rong.long@colorado.edu)) or H.A. Wu  
9       ([wuha@ustc.edu.cn](mailto:wuha@ustc.edu.cn))

10

11      **Abstract**

12       Delamination of rigid objects from an elastic substrate with finite thickness is a fundamental  
13       problem underlying applications such as marine fouling release coatings or anti-icing coatings.  
14       Most existing theoretical studies assume that delamination is driven by forces normal to the  
15       substrate surface, while in practice the delamination force may also include shear components  
16       that are parallel to the substrate surface. In this work, we consider a model system where a rigid  
17       cylindrical punch is detached from an elastic substrate under normal force, shear force or both.  
18       Our focus is to determine the pull-off force and to reveal the delamination mechanics under  
19       various geometrical and loading conditions, specifically the substrate thickness and the position  
20       and angle of the delamination force. To gain theoretical insights, we first study a plane strain  
21       model where a long rigid strip is adhered to an elastic half-space, and obtain an analytical  
22       solution revealing how the pull-off force depends on the loading position and angle. Moreover,  
23       we develop a three-dimensional finite element model to simulate the delamination of a rigid  
24       cylindrical punch from an elastic substrate with finite thickness. Three delamination modes are  
25       identified from finite element results: Mode-I crack propagation, Mode-II crack propagation,  
26       and interface cavitation. For the first two modes, we obtain empirical formulas to calculate the  
27       pull-off force using adhesion energy, substrate modulus, contact radius and substrate thickness.  
28       We also find that the analytical solution derived from the plan strain model can serve as a  
29       qualitative guide to estimate the effect of loading position and angle on the pull-off force.

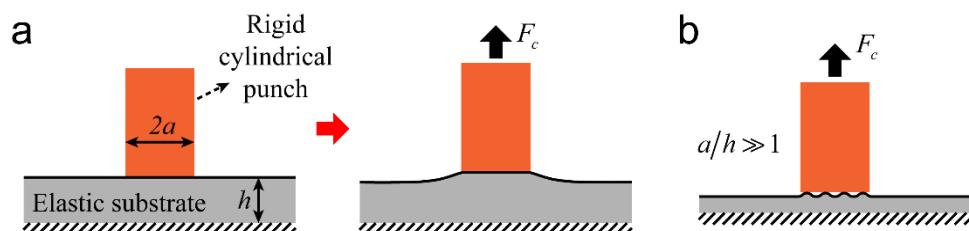
30

31      **Keywords**

32       Adhesion, delamination, pull-off force, interface fracture, finite element simulations.

34 **1. Introduction**

35 Adhesion of rigid objects on elastomeric coatings is a ubiquitous problem found in  
 36 many engineering applications. For example, marine biofouling, caused by the undesirable  
 37 attachment of marine organisms to submerged surfaces such as ship hulls, can increase the  
 38 weight of marine vessels and roughen the hull surface, leading to reduced fuel efficiency<sup>1</sup> and  
 39 increased maintenance costs<sup>2</sup>. Conventional antifouling coatings are mainly based on using  
 40 broad spectrum biocides to prevent the settlement and growth of marine organisms<sup>3,4</sup>. Serious  
 41 environment concerns have been raised towards such toxic coatings<sup>4,5</sup>, and new non-toxic  
 42 technologies to address the biofouling problem are highly desirable<sup>6-9</sup>. One approach is to  
 43 implement fouling release coatings (FRCs)<sup>6</sup> consisting of polydimethylsiloxane (PDMS) or  
 44 other soft elastomers<sup>10</sup>. Instead of preventing the attachment of fouling organisms, FRCs can  
 45 promote release of the already attached fouling organisms under external forces<sup>6</sup> due to their  
 46 low surface energy and compliance. Low adhesion strength is also desired for deicing or anti-  
 47 icing applications, i.e., to mitigate the hazardous ice accretion on aircrafts or wind turbines.  
 48 Icephobic coatings based on soft elastomers<sup>11-14</sup> have been recently developed to reduce the  
 49 force required to release ice blocks from the coating surface.



52 **Figure 1** (a) Delamination of a rigid cylindrical punch from an elastic substrate under normal  
 53 force. (b) Interface cavitation may occur in the limit of thin substrate ( $a/h \gg 1$ ).

54 In both applications described above, the force required to delaminate a rigid object,  
 55 either a barnacle or an ice block, adhered to the coating surface is an important metric for  
 56 evaluating effectiveness of the coating. Theoretical modeling of the adhesion mechanics  
 57 involved in the delamination process can enable the prediction of adhesive forces, and thus is  
 58 an important step for designing FRCs or icephobic coatings. The most widely used model

59 consists of a rigid cylindrical punch, which represents the fouling organism (e.g. barnacle) or  
 60 ice block, in adhesive contact with a soft elastic substrate (i.e., the coating layer) bonded to a  
 61 rigid backing surface (see Fig.1a). This model was analyzed in a pioneering work by Kendall<sup>15</sup>  
 62 where delamination was assumed to be driven by a force normal to the substrate. Using an  
 63 energy approach, Kendall<sup>15</sup> derived the force required to detach the punch from the substrate,  
 64 referred to as the pull-off force  $F_c$ , in two limiting cases. When the substrate (thickness:  $h$ ) is  
 65 thick in comparison to the punch radius  $a$ , the pull-off force is

$$66 \quad F_c = \pi a^2 \sqrt{\frac{8EW_{ad}}{\pi a(1-\nu^2)}}, \quad (h \gg a), \quad (1.1)$$

67 where  $E$  and  $\nu$  are the Young's modulus and Poisson's ratio of the substrate, respectively, and  
 68  $W_{ad}$  is the work of adhesion, i.e. the energy per unit area needed to separate the punch from the  
 69 substrate. On the other hand, the thin substrate limit, namely  $h \ll a$ , is more useful in practice  
 70 since most coatings fall into the limit. According to Kendall<sup>15</sup>, the pull-off force in the thin  
 71 substrate limit is

$$72 \quad F_c = \pi a^2 \sqrt{\frac{2KW_{ad}}{h}} = \pi a^2 \sqrt{\frac{2EW_{ad}}{3(1-2\nu)h}}, \quad (h \ll a), \quad (1.2)$$

73 where  $K$  is the bulk modulus. However, elastomers are typically incompressible, i.e.,  $\nu$  close  
 74 to 1/2 and  $K$  approaching infinity, for which Eq.(1.2) would predict an infinite pull-off force  
 75 and thus is not valid. To understand the origin of this limitation, we note that Eq.(1.2) was  
 76 based on the following result for the substrate compliance  $C$  proposed by Kendall<sup>15</sup>:

$$77 \quad C = \frac{\Delta}{F} = \frac{h}{\pi a^2 K} \quad (h \ll a), \quad (1.3)$$

78 where  $F$  is the force applied to the punch and  $\Delta$  is the corresponding displacement. Note that  
 79 for a flat punch with a fixed contact area and assuming linear elasticity, the mechanical response  
 80 of the substrate is linear, i.e.,  $\Delta$  is proportional to  $F$ . When the substrate is incompressible,  
 81 Eq.(1.3) implies that the substrate compliance is zero, which further leads to the singular pull-  
 82 off force in Eq.(1.2). Physically this is because when the thin substrate is confined between two  
 83 rigid surfaces (i.e., punch and backing surface), the incompressibility constraint can prevent

84 deformation of the substrate and thus diminish the compliance. This phenomenon was studied  
 85 by Lin et al.<sup>16</sup> in detail, where the compliance of thin substrate ( $h \ll a$ ) was found to be similar  
 86 to Eq.(1.3) except an additional correction factor  $(1+\nu)/[3(1-\nu)]$ . This correction factor is  
 87 equal to 1 when  $\nu=1/2$ .

88 The theoretical difficulty invoked by the bulk modulus in Eq.(1.2) was addressed by  
 89 Yang and Li<sup>17</sup> who performed a rigorous analysis for incompressible substrate ( $\nu = 1/2$ ) with  
 90 finite thickness. Specifically, they considered two different types of frictional boundary  
 91 conditions, no-slip or frictionless, at two interfaces: i) between the punch and the substrate, ii)  
 92 between the substrate and the backing surface. If the no-slip condition was assumed on both  
 93 interfaces, the compliance  $C$  in the thin substrate limit was found to be<sup>17</sup>

$$94 \quad C = \frac{2h^3}{\pi a^4 E} \quad (h \ll a). \quad (1.4)$$

95 In comparison to Eq.(1.3), the unbounded bulk modulus  $K$  is replaced by the Young's modulus  
 96  $E$  in Eq.(1.4). The low compliance due to the confinement of thin substrates is reflected in the  
 97 higher-order dependence of  $C$  on  $h/a$  (i.e.,  $C \sim h^3/a^3$  instead of  $\sim h/a$  as  $h/a \rightarrow 0$ ). The pull-off  
 98 force corresponding to Eq.(1.4) was found to be<sup>17</sup>

$$99 \quad F_c = \pi a^2 \sqrt{\frac{EW_{ad}}{2h}} \frac{a}{h}, \quad (h \ll a). \quad (1.5)$$

100 This solution predicts that  $F_c$  scales with the coating thickness  $h$  as  $F_c \sim h^{-3/2}$ , while Kendall's  
 101 theory predicts that  $F_c \sim h^{-1/2}$ . Interestingly, the latter scaling, i.e.,  $F_c \sim h^{-1/2}$ , was often observed  
 102 in experimental data with thin substrates, e.g., in adhesion experiments between plastic discs  
 103 and gelatin thin films<sup>15</sup> and between epoxy studs and silicone elastomer coatings<sup>18,19</sup>, despite  
 104 the theoretical rigorousness of the solution in Eq.(1.5). Yang and Li<sup>17</sup> showed that if the  
 105 frictionless condition is assumed on both interfaces, the pull-off force is  $F_c = \pi a^2 \sqrt{8EW_{ad}/3h}$ ,  
 106 much smaller than that in Eq.(1.5) since  $a/h \gg 1$ . This is because the frictionless interfaces  
 107 allow lateral strain in the substrate and thus can relax the confinement effect. Although the  
 108 scaling  $F_c \sim h^{-1/2}$  for frictionless interfaces is consistent with experimental observations, in  
 109 practice the coating layer is typically well bonded to the backing surface<sup>20</sup> where the no-slip

110 condition should prevail. Indeed, in this work we assume the elastic substrate is bonded  
111 perfectly to the rigid backing surface (see Fig.1a). On the punch/substrate interface, we assume  
112 strong friction and model it using a mixed-mode fracture criterion.

113 The discrepancy outlined above was reconciled by Chung and Chaudhury<sup>21</sup> who  
114 pointed out that for very thin incompressible substrates ( $h \ll a$ ), the delamination process does  
115 not initiate at the periphery of the punch, followed by an unstable interface crack propagation  
116 inward, as assumed by Kendall<sup>15</sup> and Yang and Li<sup>17</sup>. Instead, delamination initiates due to the  
117 interface cavitation instability (see Fig.1b)<sup>21–26</sup>. Interface cavitation allows local delamination  
118 within the contact area and can also relax the confinement effect for thin and incompressible  
119 substrates. In this case, the pull-off force was determined through a perturbation analysis<sup>21</sup>:

120 
$$F_c = \pi a^2 \sqrt{\frac{3.3 W_{ad} E}{h}}, \quad (h \ll a), \quad (1.6)$$

121 which gives a scaling relation  $F_c \sim h^{-1/2}$  consistent with experimental observations. Furthermore,  
122 experimental verification of the scaling relation  $F_c \sim (W_{ad} E/h)^{1/2}$  was reported in Chaudhury et  
123 al.<sup>27</sup> who used a well-defined model system to control  $E$  and  $W_{ad}$  independently and to observe  
124 interface cavitation *in situ*.

125 All studies reviewed above are based on the assumption that the delamination force is  
126 normal to the coating surface. In reality, the delamination force can come from different  
127 physical origins depending on the applications. For example, the force to release barnacles or  
128 other fouling organisms from a FRC can be provided by the hydrodynamic shear forces for a  
129 cruising ship<sup>28</sup>, and the force for ice release can come from the aerodynamic shear force for an  
130 aircraft or the centripetal force for a rotating wind turbine. In all of these scenarios, the  
131 delamination force may include components both parallel and normal to the coating surface<sup>29,30</sup>.  
132 In particular, the adhesion strength of ice on icephobic coatings is often tested under the shear  
133 mode and reported as the average shear stress at pull-off  $\tau_{ice}$ , i.e., the pull-off force divided by  
134 the contact area<sup>12–14</sup>. Although many theoretical<sup>15–17,21,26,31–33</sup> and experimental works<sup>21,23–26</sup>  
135 have been performed on the delamination mechanics under normal forces, much less work<sup>20</sup>

136 has been done for the mechanics of delamination under shear forces or combined normal and  
137 shear forces. The brief review above implies that even under normal delamination, the pull-off  
138 force is sensitive to the interface separation process. If shear force is present, the stress state of  
139 the substrate during delamination becomes inherently three-dimensional (3D), and a systematic  
140 understanding on how such 3D stress state affects the interface separation process is currently  
141 lacking.

142 The focus of this paper is on the delamination mechanics involving a rigid cylindrical  
143 punch in adhesive contact with an elastic substrate under normal and shear forces. This model  
144 system is widely used to evaluate the performance of elastomeric fouling release and icephobic  
145 coatings. The delamination of a cylindrical punch under shear forces involves 3D stress/strain  
146 states in the substrate. To gain theoretical insights towards this complex problem, we first  
147 consider a plane strain geometry in Section 2 where a rigid rectangular punch is detached from  
148 an elastic half-space (with infinite thickness), and obtain analytical solutions for the pull-off  
149 force under combined shear and normal forces. In Sections 3 and 4, we study the delamination  
150 of a rigid cylindrical punch from an elastic substrate with finite thickness using a 3D finite  
151 element (FE) model. The model is described in Section 3 while the results are presented and  
152 discussed in Section 4. Specifically, we identify the pull-off force for different substrate  
153 thickness and loading modes, based on which empirical formulas are developed. The various  
154 delamination modes revealed by the FE results and how they are related to the pull-off forces  
155 are also discussed. Conclusions are given in Section 5.

156

157 **2. Plane strain model: 2D analytical solution**

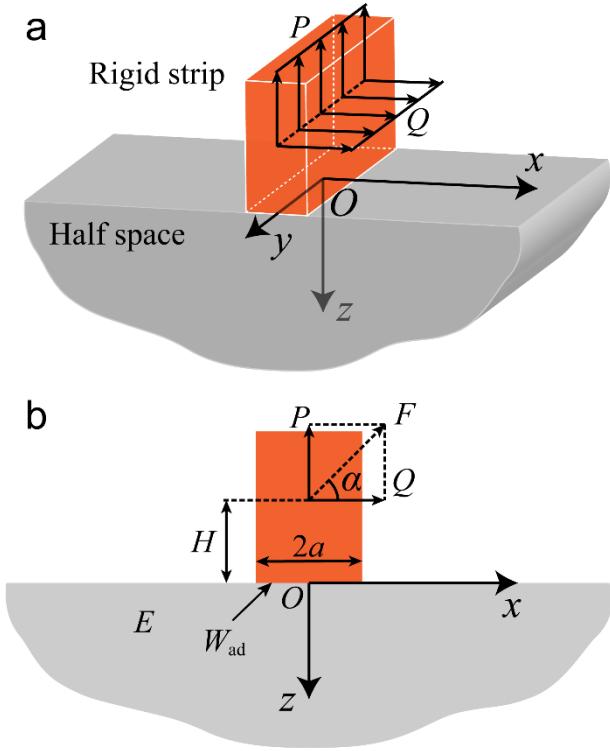
158 **2.1 Problem description**

159 In this section, we consider the delamination of a long rigid punch from an elastic half-  
160 space as shown in Fig.2a. To facilitate analysis, a Cartesian coordinate system is introduced  
161 such that the  $x$ - $y$  plane coincides with the surface of the half-space and the  $z$ -axis is directed  
162 into the half-space. The punch, assumed to be infinitely long along the  $y$ -axis, is in adhesive

163 contact with the half-space and is under line forces  $P$  and  $Q$  (unit: N/m and distributed along  
164 the  $y$ -axis) that are normal and parallel to the  $x$ - $y$  plane, respectively. The delamination process  
165 can be modeled as a two-dimensional (2D) plane strain problem in the  $x$ - $z$  plane, as shown in  
166 Fig.2b. The origin  $O$  is located at the midpoint of the contact area which occupies the region  
167 from  $x = -a$  to  $x = a$ .

168 We assume the half-space to be a linear elastic solid with Young's modulus  $E$  and  
169 Poisson's ratio  $\nu=0.5$ . The latter is motivated by the fact that most of the soft fouling release  
170 or anti-icing coatings consist of silicone elastomers (e.g. PDMS) which are  
171 incompressible<sup>6,14,34</sup>. In addition, experimental measurements suggested that the adhesion  
172 strength, defined as the average normal or shear stress on the interface when pull-off occurs, is  
173 on the order of 10-100 kPa between barnacles or ice blocks with such soft substrates<sup>6,14,28</sup>.  
174 Since the Young's modulus  $E$  for silicone elastomers is on the order of 1 MPa<sup>6,14</sup>, this range of  
175 adhesion strength implies a level of strain that is roughly 1-10%. Therefore, although nonlinear  
176 effects associated with large deformation may still be important for cases with strong adhesion,  
177 linear elasticity is a relevant assumption to the fouling release or anti-icing applications. The  
178 adhesive interaction between the punch and half-space is quantified by the adhesion energy  
179  $W_{ad}$  (unit: J/m<sup>2</sup>), defined as the energy required to separate a unit area of contact. We assume  
180 the interface adhesion to be isotropic, meaning that  $W_{ad}$  is independent of the local separation  
181 mode, e.g., along the normal or shear directions. In other words,  $W_{ad}$  is independent of the  
182 mode-mixity of the interface fracture process. As a result, there can be no slip between the  
183 punch and the elastic substrate before delamination occurs. This boundary condition is different  
184 from the experimental study of Chaudhury and Kim<sup>20</sup> on the shear induced adhesive failure  
185 between a rectangular block and a thin PDMS film where the block can slide on the PDMS  
186 film before detachment.

187



188

189 **Figure 2** Schematic illustration of a rigid punch bonded to an elastic half-space and subjected  
190 to combined shear and normal loadings: (a) three-dimensional and (b) cross-sectional views.

191

192 The two line forces  $P$  and  $Q$ , acting at the midpoint of the punch with a height  $H$  above  
193 the interface (see Fig. 2b), can be combined into a resultant force  $F$  that makes an angle  $\alpha$  with  
194 the  $x$ -axis. A positive angle  $\alpha$  implies a counterclockwise rotation from the  $x$ -axis to the  
195 direction of  $F$ . The vertical and horizontal components of  $F$ , i.e.,  $P$  and  $Q$ , will be referred to  
196 as the normal force and shear force hereafter, respectively. Simple geometrical considerations  
197 lead to the following equation:

198 
$$Q = F \cos \alpha, \quad P = F \sin \alpha. \quad (2.1)$$

199 We emphasize that  $F$ ,  $P$  and  $Q$  are “line forces” for which the unit is force per unit length. Next  
200 we derive an analytical solution illustrating how the critical force at pull-off, denoted by  $F_c$ ,  
201 depends on mechanical, interface, and geometrical parameters, i.e.,  $E$ ,  $W_{ad}$ ,  $a$ ,  $H$  and  $\alpha$ .

202 **2.2 General solution**

203 Since the punch is assumed to be rigid, either of the two edges of the contact region can

be considered as the tip of an interface crack. Whether pull-off occurs is determined by prescribing a fracture criterion at the contact edge, i.e.  $G = W_{ad}$  where  $G$  is the energy release rate of the interface crack. To evaluate the energy release rate and hence the pull-off force, we first need to determine the tractions within the contact region. As shown in Fig.3a, suppose the elastic half-space is subjected to a distributed normal pressure  $p(x)$  and tangential traction  $q(x)$  over the contact region ( $-a < x < a$ ). The surface tractions are zero outside the contact region. The displacement components on the surface of the half-space ( $z = 0$ ) due to the tractions  $p(x)$  and  $q(x)$  are given<sup>35,36</sup>

$$\bar{u}_x(x) = -\frac{2(1-\nu^2)}{\pi E} \int_{-a}^a q(s) \ln|x-s| ds - \frac{(1-2\nu)(1+\nu)}{2E} \left[ \int_{-a}^x p(s) ds - \int_x^a p(s) ds \right] + B_1, \quad (2.2)$$

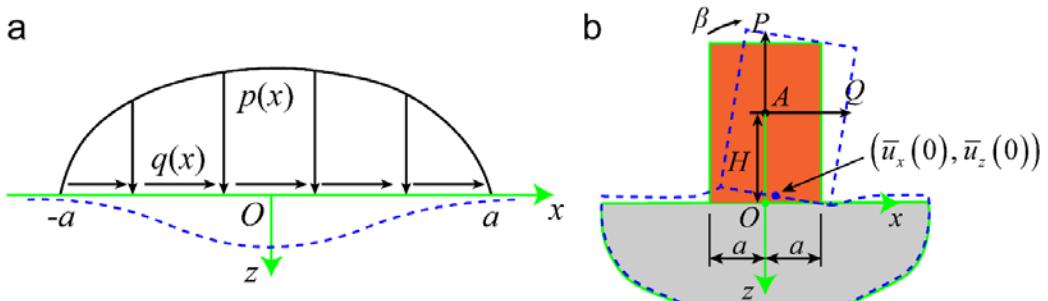
$$\bar{u}_z(z) = -\frac{2(1-\nu^2)}{\pi E} \int_{-a}^a p(s) \ln|x-s| ds + \frac{(1-2\nu)(1+\nu)}{2E} \left[ \int_{-a}^x q(s) ds - \int_x^a q(s) ds \right] + B_2, \quad (2.3)$$

where  $\bar{u}_x(x)$  and  $\bar{u}_z(x)$  are the horizontal and vertical components of the surface displacement, respectively. The constants  $B_1$  and  $B_2$  need to be determined by choosing a reference datum point on the surface. They can be removed by taking gradients of the displacement components along the surface, which gives,

$$\frac{\partial \bar{u}_x}{\partial x} = -\frac{2(1-\nu^2)}{\pi E} \int_{-a}^a \frac{q(s)}{x-s} ds - \frac{(1-2\nu)(1+\nu)}{E} p(x), \quad (2.4)$$

$$\frac{\partial \bar{u}_z}{\partial x} = -\frac{2(1-\nu^2)}{\pi E} \int_{-a}^a \frac{p(s)}{x-s} ds + \frac{(1-2\nu)(1+\nu)}{E} q(x). \quad (2.5)$$

220



221

222 **Figure 3** (a) Schematic view of the distributed shear and normal tractions acting on the top  
 223 surface of the elastic half-space. The deformed surface profile is illustrated by the blue dashed  
 224 line. (b) Cross-section of the half-space and the punch under external forces. The undeformed  
 225 and deformed states are illustrated with green solid lines and blue dashed lines, respectively.  
 226 Point *A* represents the loading position and is located at the *z*-axis with a height *H* above the  
 227 interface.

228

229 Since rigid punch is bonded to the half-space, the surface displacements within the  
 230 contact region can be specified by the motion of the punch. Specifically, the punch is expected  
 231 to rotate by a small angle  $\beta$  under combined normal and shear forces (see Fig.3b), which leads  
 232 to the following expressions for surface displacements,

233 
$$\bar{u}_x(x) = \bar{u}_x(0) - x(1 - \cos \beta), \quad (2.6)$$

234 
$$\bar{u}_z(x) = \bar{u}_z(0) + x \sin \beta, \quad (2.7)$$

235 where  $\bar{u}_x(0)$  and  $\bar{u}_z(0)$  are surface displacement components at *O* where  $x=0$ . Substituting  
 236 Eqs.(2.6) and (2.7) to Eqs.(2.4) and (2.5) will allow us to solve the surface tractions  $p(x)$  and  
 237  $q(x)$  from a set of coupled integral equations. When the incompressibility of the half-space, i.e.,  
 238  $\nu=0.5$ , is assumed, the normal and shear components are decoupled, leading to the following  
 239 equations for  $p(x)$  and  $q(x)$ :

240 
$$\frac{\partial \bar{u}_x}{\partial x} = 1 - \cos \beta = -\frac{3}{2\pi E} \int_{-a}^a \frac{q(s)}{x-s} ds, \quad (2.8)$$

241 
$$\frac{\partial \bar{u}_z}{\partial x} = \sin \beta = -\frac{3}{2\pi E} \int_{-a}^a \frac{p(s)}{x-s} ds. \quad (2.9)$$

242 Moreover, equilibrium of the rigid punch implies the following force and moment balance  
 243 equations:

244 
$$Q = \int_{-a}^a q(x) dx, \quad (2.10)$$

245 
$$P = - \int_{-a}^a p(x) dx, \quad (2.11)$$

246  $QH = \int_{-a}^a xp(x)dx.$  (2.12)

247 Note that Eq.(2.12) results from the moment balance of the rigid punch about the point  $O$ .  
 248 Combing Eqs.(2.8)-(2.11), we can obtain the following solutions for the surface tractions:

249  $q(x) = -\frac{2E(1-\cos\beta)}{3} \frac{x}{(a^2-x^2)^{1/2}} + \frac{Q}{\pi(a^2-x^2)^{1/2}}, -a \leq x \leq a,$  (2.13)

250  $p(x) = \frac{2E\sin\beta}{3} \frac{x}{(a^2-x^2)^{1/2}} - \frac{P}{\pi(a^2-x^2)^{1/2}}, -a \leq x \leq a.$  (2.14)

251 The angle of rotation  $\beta$  can be solved by substituting Eq.(2.14) into Eq.(2.12), which gives

252  $QH = \frac{\pi E a^2}{3} \sin \beta.$  (2.15)

253 Next we denote normal and shear stress components exposed on the surface of the  
 254 elastic half-space as  $\bar{\sigma}_z(x) \equiv \sigma_z(x, z=0)$  and  $\bar{\tau}_{xz}(x) \equiv \tau_{xz}(x, z=0).$  Using Eqs.(2.13)-(2.14),  
 255 we obtain

256  $\bar{\sigma}_z(x) = -p(x) = -\frac{2QH}{\pi a^2} \frac{x}{(a^2-x^2)^{1/2}} + \frac{P}{\pi(a^2-x^2)^{1/2}}, -a \leq x \leq a,$  (2.16)

257  $\bar{\tau}_{xz}(x) = -q(x) = -\frac{Q}{\pi(a^2-x^2)^{1/2}}, -a \leq x \leq a,$  (2.17)

258 where we have applied the infinitesimal deformation assumption that  $\beta \ll 1$  and retained only  
 259 the first order terms of  $\beta.$  Specifically, the first term of  $q(x)$  is proportional to  $(1-\cos\beta) \sim \beta^2$  and  
 260 thus is neglected in Eq.(2.17), while the coefficient  $2E\sin\beta/3$  in the first term of  $p(x)$  is  
 261 substituted by  $2QH/\pi a^2$  using Eq.(2.15). The two equations above show that both the normal  
 262 and shear stress components exhibit a square root singularity at the contact edges  $x = \pm a.$  This  
 263 is because the contact edge is equivalent to the tip of an interface crack between the rigid punch  
 264 and the elastic half-space. Note that the stress field near the tip of an interface crack often  
 265 exhibits an oscillatory singularity<sup>37</sup>, which is not present here because the substrate is

266 incompressible ( $\nu=1/2$ ) and the oscillation is removed<sup>16</sup>. The Mode I and II stress intensity  
 267 factors, denoted by  $K_I$  and  $K_{II}$  respectively, at the two bonding edges  $x = \pm a$  can be calculated  
 268 as follows:

$$269 \quad K_I^{(a)} = \lim_{x \rightarrow a} \bar{\sigma}_z(x) \sqrt{2\pi(a-x)} = -2 \frac{Q}{\sqrt{\pi a}} \frac{H}{a} + \frac{P}{\sqrt{\pi a}}, \quad (2.18)$$

$$270 \quad K_{II}^{(a)} = \lim_{x \rightarrow a} \bar{\tau}_{xz}(x) \sqrt{2\pi(a-x)} = -\frac{Q}{\sqrt{\pi a}}, \quad (2.19)$$

$$271 \quad K_I^{(-a)} = \lim_{x \rightarrow (-a)} \bar{\sigma}_z(x) \sqrt{2\pi(a+x)} = 2 \frac{Q}{\sqrt{\pi a}} \frac{H}{a} + \frac{P}{\sqrt{\pi a}}, \quad (2.20)$$

$$272 \quad K_{II}^{(-a)} = \lim_{x \rightarrow (-a)} \bar{\tau}_{xz}(x) \sqrt{2\pi(a+x)} = -\frac{Q}{\sqrt{\pi a}}. \quad (2.21)$$

273 These expressions of the  $K_I$  and  $K_{II}$  allow us to evaluate the energy release rate at the two  
 274 contact edges based on the following equation<sup>16</sup>:

$$275 \quad G = \left( \frac{1-\nu^2}{2E} \right) (K_I^2 + K_{II}^2). \quad (2.22)$$

276 Using Eqs.(2.18)–(2.21) and setting the Poisson's ratio  $\nu=0.5$ , we obtain the following results  
 277 for the strain energy release rates at  $x = \pm a$ :

$$278 \quad G^{(-a)} = \frac{3}{8\pi a E} \left( \left( \frac{2QH}{a} + P \right)^2 + Q^2 \right) \text{ and } G^{(a)} = \frac{3}{8\pi a E} \left( \left( -\frac{2QH}{a} + P \right)^2 + Q^2 \right). \quad (2.23)$$

279 Without loss of generality, we assume both  $P$  and  $Q$  to be positive, which implies that  $G^{(-a)} >$   
 280  $G^{(a)}$ . As a result, delamination should initiate at the left edge where  $x = -a$ . Recalling that the  
 281 interfacial adhesion energy is by  $W_{ad}$  and the substrate is elastic, the onset of delamination  
 282 occurs when  $G^{(-a)} = W_{ad}$ . Using Eq.(2.1), we obtain the following equation for the critical force  
 283  $F_c$  to initiate delamination:

$$284 \quad F_c = \sqrt{\frac{2\pi a E W_{ad}}{3 \left( \frac{H^2}{a^2} \cos^2 \alpha + \frac{H}{a} \cos \alpha \sin \alpha + \frac{1}{4} \right)}}, \quad (2.24)$$

285 where  $\alpha$  is the angle of the combined force  $F$  in Fig.2b ( $0 \leq \alpha \leq \pi/2$ ). Note that  $F_c$  is a  
 286 monotonically decreasing function of the contact width  $a$ . This implies that the delamination  
 287 process is unstable under force control once it initiates at the contact edge. Therefore, the  $F_c$  in  
 288 Eq.(2.24) is the pull-off force we are looking for.

289

290 **2.3 Pull-off force**

291 Although we have assumed that the delamination process is independent of the local  
 292 fracture modes at the contact edge, i.e.,  $W_{ad}$  is mode-independent, the pull-off force does  
 293 depend on the mode of delamination, represented by the direction  $\alpha$  and height  $H$  of the  
 294 delamination force  $F$ . For example, under normal separation where  $\alpha=\pi/2$ , the pull-off force,  
 295 denoted by  $P_c$ , is

$$296 \quad P_c = \sqrt{\frac{8\pi a E W_{ad}}{3}}. \quad (2.25)$$

297 On the other hand, under shear delamination where angle  $\alpha=0$ , the pull-off force  $Q_c$  given by  
 298 Eq.(2.24) is

$$299 \quad Q_c = \sqrt{\frac{8\pi a E W_{ad}}{3\left(4\frac{H^2}{a^2}+1\right)}}. \quad (2.26)$$

300 Interestingly, the shear pull-off force  $Q_c$  is smaller than the normal pull-off force  $P_c$  except  
 301 when  $H=0$ .

302 To further illustrate the effect of  $H$  and  $\alpha$  on the pull-off force  $F_c$ , we use the normal  
 303 pull-off force  $P_c$  as the benchmark to define the following normalized pull-off force:

$$304 \quad \bar{F}_c = \frac{F_c}{P_c} = \sqrt{\frac{1}{4\frac{H^2}{a^2}\cos^2\alpha + 4\frac{H}{a}\cos\alpha\sin\alpha + 1}}. \quad (2.27)$$

305 The fact that  $\bar{F}_c \leq 1$  implies that normal delamination requires the largest pull-off force. Figure

306 4 shows a 3D surface plot of the normalized pull-off force versus  $\alpha$  and  $H$ , as well as the  
 307 dependence of pull-off force on  $\alpha$  for different values of  $H/a$ .

308 We first observe that  $\bar{F}_c$  is a monotonically decreasing function with increasing  $H$ .  
 309 This is because the shear component  $Q$  of the delamination force can cause the punch to rotate  
 310 in the clockwise direction. Larger  $H$  results in a larger torque by  $Q$  and hence larger rotation.  
 311 Such rotation can modify the local mixed-mode fracture conditions at the two contact edges ( $x$   
 312  $= \pm a$ ) by developing an extra normal stress at the interface that decays from the left contact  
 313 edge to the right one. This effect, under a special case of  $\alpha = 0$  (i.e., shear delamination), has  
 314 been discussed in Chaudhury and Kim<sup>20</sup>. It can create an uneven distribution of energy release  
 315 rate (see Eq.(2.23)), and cause the delamination to initiate only at the left contact edge ( $x = -a$ ).  
 316 In contrast, under a pure normal force, both contact edges are subjected to the same energy  
 317 release rate, and thus will experience delamination simultaneously. This mechanism causes  $F_c$   
 318 to decrease with increasing  $H$ . When  $H=0$ ,  $\bar{F}_c=1$  regardless what the angle  $\alpha$  is. In this  
 319 particular case, the punch does not rotate, and the energy release rates at both contact edges are  
 320 equal to each other. The angle  $\alpha$  can only affect the mode-mixity conditions at the two contact  
 321 edges. However, since we have assumed  $W_{ad}$  is independent of the local mixed-mode condition,  
 322 the pull-force becomes independent of  $\alpha$  and is equal to the normal pull-off force  $P_c$ .

323 Next we discuss the effect of angle  $\alpha$ . The normalized pull-off force  $\bar{F}_c$  is not a  
 324 monotonic function of  $\alpha$ . Indeed, we find that  $\bar{F}_c$  reaches the following minimum:

$$325 \quad \bar{F}_{c\_min} = \sqrt{1 + \left(\frac{H}{a}\right)^2} - \frac{H}{a}, \quad (2.28)$$

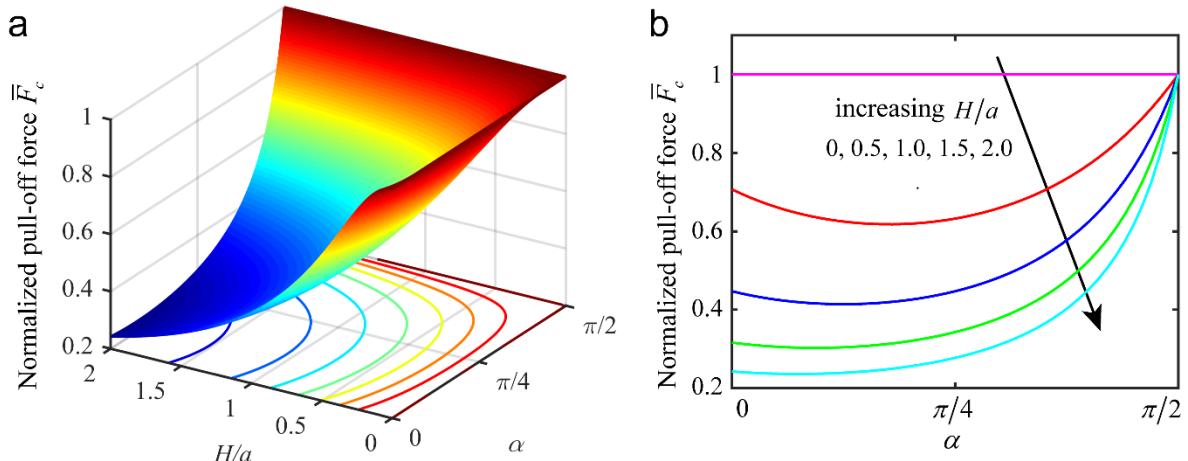
326 when  $\alpha$  is equal to  $\alpha_m$  below:

$$327 \quad \alpha_m = \tan^{-1} \left( \sqrt{1 + \left(\frac{H}{a}\right)^2} - \frac{H}{a} \right). \quad (2.29)$$

328 This implies that for a given value of  $H/a$ , the direction of the delamination force can be  
 329 adjusted to minimize the pull-off force according to Eq.(2.29). For example, the angle  $\alpha_m$

330 required to minimize the pull-off force would decrease to 0 as  $H/a$  approaches infinity, meaning  
 331 that in the limit of large  $H/a$ , the shear delamination mode ( $\alpha=0$ ) should be applied to minimize  
 332 the pull-off force.

333



334

335 **Figure 4** (a) Contour plot of the normalized pull-off force versus  $H/a$  and  $\alpha$ . (b) The  
 336 dependence of the pull-off force on  $\alpha$  for different values of  $H/a$ .

337

338 In this section we have used a 2D plane strain model to understand how the pull-off  
 339 force depends on the mode of delamination, represented by the angle  $\alpha$  and loading position  $H$ .  
 340 The analytical solutions obtained in this section can be potentially used to guide the design of  
 341 delamination modes to enhance or reduce the pull-off force. Moreover, the fundamental  
 342 insights established in this section will also help us understand the 3D delamination behaviors  
 343 to be considered in Section 3 and 4.

344

### 345 **3. Finite element model: 3D simulations**

#### 346 **3.1 Model description**

347 In this section, we consider the delamination process of a rigid cylindrical punch from  
 348 an elastic substrate under normal and shear forces, as schematically shown in Fig.5a. Here we  
 349 assume the substrate has a finite thickness  $h$  and is bonded to a rigid backing surface. This is

350 motivated by the fact that in practice the FRC or anti-icing coatings are usually supported by a  
351 stiff surface, and the coating thickness is comparable or even smaller than dimensions of the  
352 adhering objects (e.g. barnacles or ice blocks). Therefore, the half-space assumption may not  
353 always be satisfied for practical applications and the effect of finite substrate thickness needs  
354 to be accounted for.

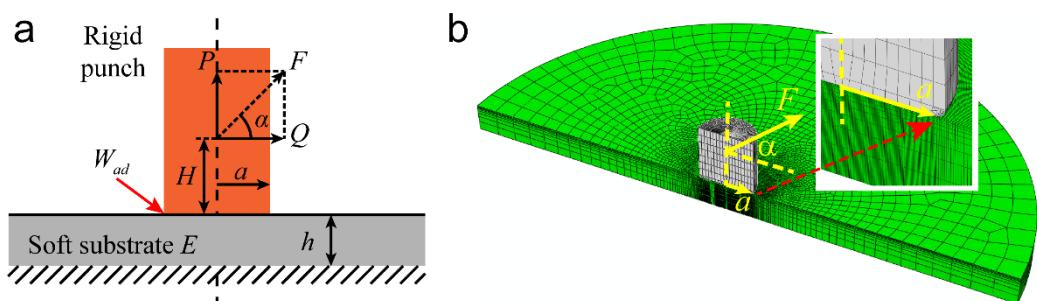
355 The existing solutions for pull-off forces reviewed in Section 1 focused on normal  
356 delamination, where one can take advantage of axisymmetry to solve the interface fracture  
357 problem into a 2D domain. In our case, the combined normal and shear forces would induce a  
358 3D stress and strain fields in the elastic substrate, which is difficult to solve analytically.  
359 Therefore, we resort to numerical simulations and build an FE model in ABAQUS (v6.14,  
360 Simulia Inc, Providence, RI) to calculate the pull-off force. Figure 5b shows the finite element  
361 model wherein the symmetry allows us to simulate a half of the cylinder and substrate. The  
362 cylindrical punch is modeled as a discrete rigid body, and the soft substrate with Young's  
363 modulus  $E$  is modeled as an incompressible neo-Hookean solid. The substrate is meshed into  
364 263080 C3D8RH elements and 920 C3D6H elements. We have performed mesh convergence  
365 test (see Appendix 1) to ensure mesh independence of our results. The adhesive interaction on  
366 the contacting interface is described by a cohesive zone model<sup>38</sup>, which will be discussed in  
367 Section 3.2. Regarding boundary conditions, the circumferential and bottom surfaces of the  
368 substrate are fixed, and the lateral cross-section exposed by the symmetry cut is subjected to  
369 the symmetry boundary condition, i.e., no normal displacements and no shear tractions. The  
370 delamination force is applied to the cylindrical punch, and the point of action is located on the  
371 central axis of the punch with a height  $H$  above the substrate surface (see Fig.5b). To make sure  
372 that the delamination process is not affected by the circumferential boundary of the substrate,  
373 the radius of the substrate is set to be 10 times of the punch radius.

374 Since the punch may become tilted forward due to the shear force during delamination,  
375 its front edge can be pressed into the substrate, which may cause severe local stress  
376 concentrations and mesh distortion. To resolve this problem, we introduce a fillet of  $0.1r$ , where  
377  $r$  denotes the punch radius, at the bottom edge of the punch (see inset of Fig.5b). As a result,

378 the contact radius is  $0.9r$  rather than  $r$ . We will denote the contact radius as  $a$  to maintain  
 379 consistency with notations of the plane strain model. Since the punch is rigid, introduction of  
 380 the fillet does not affect the local stress states near the delamination site (see inset of Fig.5a).  
 381 Therefore, the delamination behavior and pull-off force is independent of the fillet.

382 We use the dynamic implicit solver in ABAQUS to capture the rapid unloading once  
 383 the delamination process is initiated. In addition, we adopt displacement-controlled loading to  
 384 stabilize the simulation, namely that a prescribed displacement along the desired angle  $\alpha$ ,  
 385 instead of a prescribed force, is applied to the punch, while other degrees of freedom of the  
 386 rigid punch (e.g. rotation) are unconstrained. A local Cartesian coordinate system with one axis  
 387 aligned with the loading direction is used to ensure that the direction of delamination force is  
 388 kept constant throughout the entire simulation. The main advantage of the displacement  
 389 controlled loading is that it can capture pull-off force and the subsequent delamination much  
 390 more efficiently than the force controlled loading.

391



392

393 **Figure 5** (a) Schematic of a cross-section of a cylindrical rigid punch adhering to a finite-  
 394 thickness soft substrate with the bottom surface being fixed. The applied force and thus the  
 395 deformation of the substrate are non-axisymmetric. (b) 3D mesh of FE model of the punch and  
 396 the substrate with a representative thickness. The symmetry of the model allows us to perform  
 397 the simulation by using the half-model. The inset shows the fillet of the punch. The cylinder  
 398 radius subtracted by the fillet radius is defined as  $a$ .

399

### 400 **3.2 Cohesive zone model**

401 The cohesive zone model is defined by prescribing a relation between the mechanical

402 traction on the interface and the relative separation between the two contacting surfaces.  
 403 Complete interface failure occurs when the maximum separation  $\delta_f$  is reached and the traction  
 404 reduces to 0. The energy required to achieve complete interface failure of a unit contacting area  
 405 is defined as the adhesion energy  $W_{ad}$ . In general, the global delamination behavior, e.g. the  
 406 pull-off force, is insensitive to the detailed shape of the traction-separation curve, as long as  
 407  $W_{ad}$  is kept constant and the maximum separation  $\delta_f$  is much smaller than the characteristic  
 408 length scales of the contacting region<sup>38</sup>. Therefore, we adopt a simple bi-linear traction-  
 409 separation law for our cohesive zone model, which is illustrated in Fig.6. The adhesion energy  
 410  $W_{ad}$  is equal to the area underneath the traction-separation curve. To validate our cohesive zone  
 411 model, we tested the effects of different parameters for the cohesive zone model. Using normal  
 412 delamination as a benchmark example, we find that the pull-off forces predicted by our  
 413 simulations based on the following cohesive zone parameters agree well with those given by  
 414 Yang and Li<sup>17</sup>:

$$415 \quad \delta_0 = \frac{1}{2} \delta_f = \frac{W_{ad}}{\sigma_{\max}}, \quad \frac{\delta_f}{a} = 0.01. \quad (3.1)$$

416 These parameters will be used throughout our simulations. Note that the element size within  
 417 the contacting area is selected to be smaller than  $\delta_f$  to ensure mesh convergence within the  
 418 cohesive zone.

419 To identify the extent of partial interface separation in the simulations, we use a damage  
 420 variable  $D$  defined by the cohesive zone model. This variable  $D$  ranges from 0 to 1, with 1  
 421 representing complete separation. Let  $\delta_m$  represent the maximum separation of the cohesive  
 422 zone attained in the loading history. When  $\delta_m$  is less than  $\delta_0$  (i.e., the separation at peak stress;  
 423 see Fig.6), damage does not occur and  $D = 0$ . Damage is initiated once  $\delta_m > \delta_0$  and  $D$  is  
 424 calculated according to the linear evolution model:

$$425 \quad D = \frac{\delta_f (\delta_m - \delta_0)}{\delta_m (\delta_f - \delta_0)}. \quad (3.2)$$

426 In a 3D model, the traction-separation relation needs to be defined along three directions: one

427 normal to the interface and the other two tangential to the interface. Here, we assume an  
 428 isotropic interfacial behavior with respect to three directions. The damage initiation is governed  
 429 by a quadratic stress criterion which requires

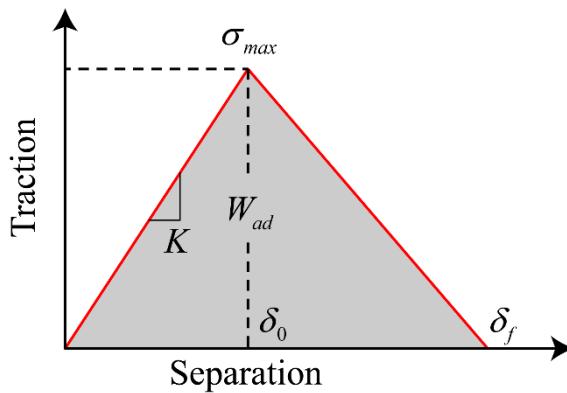
$$430 \quad \left( \frac{\sigma_n}{\sigma_{\max}} \right)^2 + \left( \frac{\sigma_s}{\sigma_{\max}} \right)^2 + \left( \frac{\sigma_t}{\sigma_{\max}} \right)^2 = 1, \quad (3.3)$$

431 with  $\sigma_n$ ,  $\sigma_s$ ,  $\sigma_t$  representing the normal and the two shear tractions, respectively, and the damage  
 432 behavior in mixed mode are described by the built-in power law with the exponent of 2, i.e.,

$$433 \quad \left( \frac{W_n}{W_{ad}} \right)^2 + \left( \frac{W_s}{W_{ad}} \right)^2 + \left( \frac{W_t}{W_{ad}} \right)^2 = 1, \quad (3.4)$$

434 where  $W_n$ ,  $W_s$  and  $W_t$  represent the work done by the normal and the two shear adhesive stresses,  
 435 respectively.

436



437

438 **Figure 6** An example of the traction-separation law for the cohesive zone model.

439

440 **3.3 Model parameters and dimensional analysis**

441 The FE model described above involves a number of material and geometrical  
 442 parameters: substrate Young's modulus  $E$ , substrate thickness  $h$ , contact radius  $a$ , height of  
 443 delamination force  $H$ , angle of delamination force  $\alpha$ , and adhesion energy  $W_{ad}$ . Here we  
 444 perform a dimensional analysis to reduce the number of independent parameters. First, we use  
 445 the  $\pi$  theorem to get

446 
$$\frac{F_c}{Ea^2} = g\left(\frac{W_{ad}}{Ea}, \frac{a}{h}, \frac{a}{H}, \alpha\right), \quad (3.5)$$

447 where  $F_c$  is the pull-off force and  $g$  is an unknown function. To further simplify this equation,  
 448 we assume infinitesimal deformation so that the elastic substrate can be considered as a linear  
 449 system represented by a compliance  $C$  with

450 
$$\Delta = CF, \quad (3.6)$$

451 where  $F$  is the force applied to the punch and  $\Delta$  is the corresponding displacement. Following  
 452 Kendall<sup>15</sup>, we can derive the following expression for the total potential energy of the system  
 453 (punch + substrate):

454 
$$U_T = \frac{1}{2}F\Delta - F\Delta - AW_{ad} = -\frac{CF^2}{2} - AW_{ad}, \quad (3.7)$$

455 where  $A$  is the contact area. Initiation of delamination requires the following equation to be  
 456 satisfied:

457 
$$\frac{\partial U_T}{\partial A} = -\frac{F_c^2}{2} \frac{\partial C}{\partial A} - W_{ad} = 0. \quad (3.8)$$

458 Since  $\partial C/\partial A$  is independent of the adhesion energy  $W_{ad}$ , Eq.(3.8) suggests that the pull-off force  
 459  $F_c$  scales with  $W_{ad}^{1/2}$ . Using this scaling and Eq.(3.5), we can write

460 
$$\frac{F_c}{Ea^2} = \sqrt{\frac{W_{ad}}{Ea}} g^*\left(\frac{a}{h}, \frac{H}{a}, \alpha\right) \quad \text{or} \quad F_c = \sqrt{Ea^3 W_{ad}} g^*\left(\frac{a}{h}, \frac{H}{a}, \alpha\right). \quad (3.9)$$

461 The scaling relation in Eq.(3.9) will be used to guide our interpretation of the pull-off force  
 462 data in Section 4.

463

464 **4. Results and Discussions**

465 In this section we present the FE results for the pull-off force  $F_c$  under various  
 466 geometric and loading conditions and, more importantly, the adhesion mechanics underlying  
 467 the pull-off forces. Eq.(3.9) motivates us to define a normalized pull-off force to reduce the

468 number of independent parameters. Similar to the plane strain case in Section 2, we use the  
 469 pull-off force  $F_{c0}$  for an elastic half-space ( $a/h = 0$ ) under normal separation ( $\alpha = \pi/2$ ) as the  
 470 reference. According to Eq.(1.1)<sup>15</sup>,  $F_{c0}$  for an incompressible substrate is

$$471 \quad F_{c0} = \sqrt{\frac{32\pi Ea^3 W_{ad}}{3}}, \quad (4.1)$$

472 which yields the following definition of normalized pull-off force

$$473 \quad \bar{F}_c = \frac{F_c}{F_{c0}} = \sqrt{\frac{3}{32\pi Ea^3 W_{ad}}} F_c = \sqrt{\frac{3}{32\pi}} g^* \left( \frac{a}{h}, \frac{H}{a}, \alpha \right). \quad (4.2)$$

474 The three independent parameters are:  $a/h$  representing the substrate thickness,  $H/a$   
 475 representing the location of loading point, and  $\alpha$  representing the angle of delamination force.

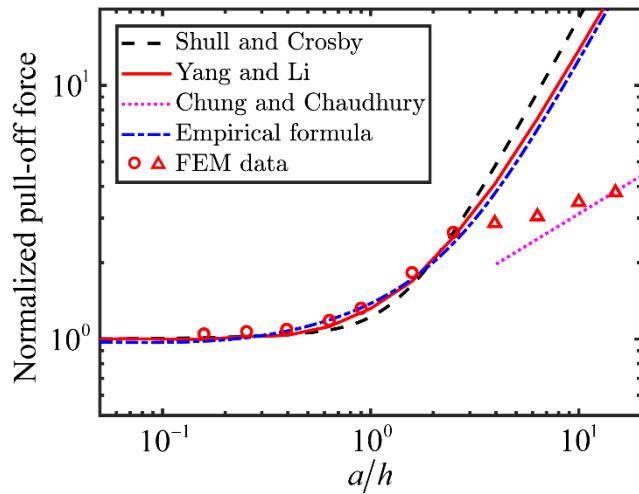
#### 476 **4.1 Normal separation**

477 We first consider the case of normal separation as a benchmark to validate our FE model,  
 478 since it has been solved in a number of previous theoretical analyses<sup>15,17,21,31</sup>. For normal  
 479 separation,  $\alpha = \pi/2$  and the delamination process is independent of the height  $H$  of loading  
 480 point. Therefore,  $\bar{F}_c$  is only a function of  $a/h$ . Figure 7 shows the FE results (symbols) of  $\bar{F}_c$   
 481 under normal separation for different values of  $a/h$ . Also plotted in Fig.7 are existing analytical  
 482 solutions in the literature. Specifically, Yang and Li<sup>17</sup> performed a rigorous analysis based on  
 483 which numerical solutions of the pull-off force were obtained for different substrate thicknesses  
 484 as showed by the red solid line. On the other hand, Shull and Crosby<sup>31</sup> first used FE simulations  
 485 to obtain an empirical expression for the compliance of the punch/substrate system, based on  
 486 which the pull-off force is determined by evaluating the energy release rate and setting it equal  
 487 to the adhesion energy  $W_{ad}$ . This resulted in an analytical formula for the pull-off force as  
 488 follows:

$$489 \quad F_c = 6.70 \sqrt{Ea^3 W_{ad}} \left[ \frac{0.75 + (a/h) + (a/h)^3}{(0.75 + 2(a/h) + 4(a/h)^3)^{1/2}} \right]. \quad (4.3)$$

490 Note that both solutions (i.e., Yang and Li<sup>17</sup> and Shull and Crosby<sup>31</sup>) adopted for comparison  
 491 here are based on the assumption of no-slip interface between the punch and the substrate and  
 492 between the substrate and the backing surface. As described in Section 1, Yang and Li<sup>17</sup> also  
 493 considered other cases of frictional boundary conditions. These solutions are not discussed here  
 494 since they are based on different interface conditions from our FEM model.

495



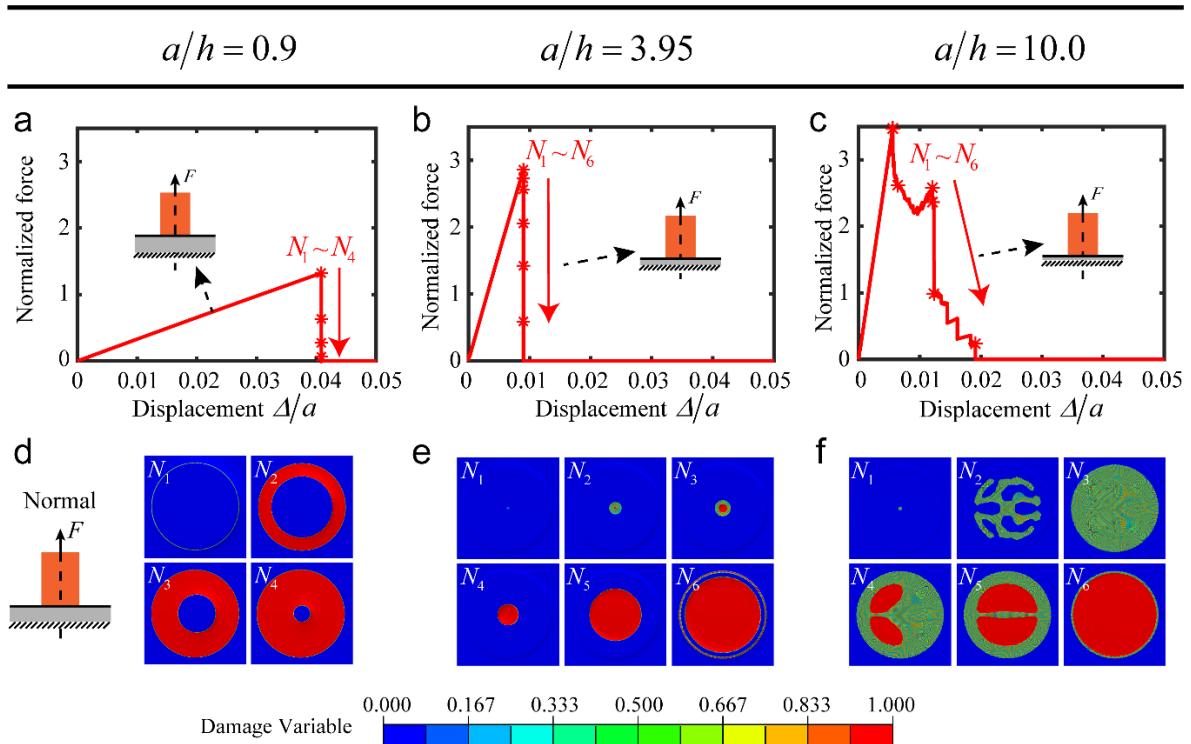
496  
 497 **Figure 7** Normalized pull-off force versus  $a/h$  under normal separation. The symbols (circles:  
 498 Mode-I crack propagation; triangles: interface cavitation) represent our FE results. The black  
 499 dashed line is obtained by the empirical solution of Shull and Crosby<sup>31</sup> and the red solid line is  
 500 obtained by the numerical solution of Yang and Li<sup>17</sup>. Our empirical formula, Eq.(4.4), is plotted  
 501 as the blue dash-dot line. The magenta dotted line illustrates the solution of Chung and  
 502 Chaudhury<sup>21</sup> which accounts for interface cavitation with thin substrates.

503

504 As can be seen in Fig.7, both Eq.(4.3) by Shull and Crosby<sup>31</sup> and the numerical solutions  
 505 by Yang and Li<sup>17</sup> agree well with our FE results for substrate with large to moderate thickness  
 506 ( $a/h \leq 2.5$ ). Interestingly, for thin substrate ( $a/h > 2.5$ ), our FE results are significantly lower  
 507 than the theoretical predictions of either Yang and Li<sup>17</sup> or Shull and Crosby<sup>31</sup>. This is because  
 508 in both of these studies, the delamination is assumed to initiate at the periphery of the circular  
 509 contact region, followed by an unstable Mode-I crack propagation inward, consistent with what  
 510 we observed in FE simulations for thick substrates. An example where  $a/h = 0.9$  is illustrated  
 511 in Fig.8d, where the color map of the interface damage variable  $D$  (see Eq.(3.2)) is plotted to

512 illustrate the evolution of delaminated region. Recall that  $D$  ranges from 0 to 1, with 1 denoting  
 513 complete delamination. The pull-off force occurs when delamination started at the outer edge  
 514 of the contact area (marked by “ $N_1$ ” in Fig.8a and 8d). In contrast, for thinner substrates ( $a/h$   
 515  $\geq 3.95$ ), the delamination does not initiate at the periphery but rather occurs through interface  
 516 cavitation. Specifically, when  $a/h = 3.95$ , Fig.8e shows that delamination first occurs in the  
 517 center of the contact area and then propagates outward. When the substrate thickness is further  
 518 reduced ( $a/h = 10$ ), multiple cavities may first be formed within the contact area, and then grow  
 519 into a larger delaminated region (see Fig.8f). These behaviors found in our simulation results  
 520 are consistent with the three normal delamination modes, i.e. edge crack propagation, internal  
 521 crack propagation and interface cavitation, experimentally observed in Crosby et al.<sup>23</sup> and  
 522 Webber et al.<sup>24</sup>.

523



524

525 **Figure 8** Delamination behavior under normal separation with different  $a/h$ . (a-c) The force-  
 526 displacement curves during delamination. (d-f) Evolution of the interface damage variable  $D$   
 527 (shown in the top view of the substrate surface) to quantify local interface delamination: (d)  
 528  $a/h = 0.9$ , (e)  $a/h = 3.95$ , (f)  $a/h = 10.0$ .

529 The underlying physical mechanism of interface cavitation has been studied in the  
 530 literature<sup>16,26,39</sup>. Briefly, the incompressibility of the substrate, together with the small thickness  
 531 and fixed boundary condition at the bottom of the substrate, can lead to a hydrostatic tensile  
 532 stress state within the contact area<sup>16</sup>. In addition, as  $a/h$  approaches infinity and assuming  
 533 perfectly bonded interface, the normal stress within the contact area was found to be<sup>33,39,40</sup>:  
 534  $\sigma_{zz}(r) = (2F/\pi a^2)(1-r^2/a^2)$  based on a first order asymptotic analysis<sup>40</sup>. This solution indicates  
 535 that the normal stress decays as one moves away from the center of the contact region. However,  
 536 near the contact edge there exists a boundary layer where the normal stress exhibits a singularity,  
 537 because the contact edge is equivalent to the tip of an interface crack between the rigid punch  
 538 and the elastic substrate<sup>16</sup>. This would lead to a steep upturn of the normal stress within the  
 539 boundary layer. The size of the boundary layer is bounded by the substrate thickness  $h$  and thus  
 540 should decrease as the substrate becomes thinner<sup>41</sup>. Note that a cohesive zone model is  
 541 implemented at the punch/substrate interface in our FEM simulations. For thin substrates (i.e.,  
 542 large  $a/h$ ), the small boundary layer at the contact region is covered by the cohesive zone, which  
 543 regulates the theoretically predicted stress singularity by allowing partial interface separation.  
 544 The cohesive zone model may also affect the normal stress distribution by introducing an  
 545 additional interface compliance<sup>42</sup>. Nevertheless, the increased normal stress towards the center  
 546 of the contact region as the substrate becomes thinner tends to promote a surface instability  
 547 that causes local delamination and hence interface cavitation<sup>21,27,43</sup>. To distinguish the two  
 548 delamination modes, i.e., unstable crack propagation or interface cavitation, in Fig.7 we use  
 549 triangular symbols to represent the FE data of pull-off force for cases with interface cavitation  
 550 while circular symbols for cases where delamination is dominated by Mode-I crack  
 551 propagation. Chung and Chaudhury<sup>21</sup> derived a solution for the pull-off force that accounts for  
 552 interface cavitation through an energy minimization method. This solution, shown in Eq.(1.6)  
 553 and plotted in Fig.7, is consistent with our FE data as the substrate becomes thinner (or  $a/h$   
 554 increases). However, our FE data do exhibit a slightly different scaling relation between  $\bar{F}_c$   
 555 and  $a/h$  from that predicted by Chung and Chaudhury<sup>21</sup>. We emphasize that since interface  
 556 cavitation involves a surface instability of the substrate, the development of cavitation may be

557 sensitive to the imperfections within the FE model. In our model, imperfections were not  
 558 deliberately seeded into the simulations, but rather came from the geometric irregularities of  
 559 the mesh. A systematic study on the effects of imperfections is required to achieve quantitative  
 560 conclusions on the pull-off force with interface cavitation, which is beyond the scope of this  
 561 paper. Therefore, our focus will be placed on cases where delamination is dominated by  
 562 interface crack propagation.

563 Finally, we modify Eq.(3.4) to the following form to obtain a more accurate formula  
 564 for the normal pull-off force:

$$565 \quad F_c = C_1 \sqrt{Ea^3 W_{ad}} \left[ \frac{0.75 + C_2(a/h) + C_3(a/h)^3}{(0.75 + C_4(a/h) + C_5(a/h)^3)^{1/2}} \right], \quad (4.4)$$

566 where  $C_1$  to  $C_5$  are constant coefficients. To ensure that Eq.(4.4) recovers Eq.(4.1) when  $a/h \ll$   
 567 1 and Eq.(1.5) when  $a/h \gg 1$ , we impose two constraints:  $C_1 = 8\sqrt{2\pi}/3$  and  
 568  $C_1 C_3 / \sqrt{C_5} = \pi / \sqrt{2}$ . By fitting Eq.(4.4) to the numerical results of Yang and Li<sup>17</sup> (red solid  
 569 line in Fig.7), we found that  $C_2 = 2.2002$ ,  $C_3 = 0.2684$ ,  $C_4 = 5.8186$ , and  $C_5 = 0.7427$  (also  
 570 summarized in Table 1). In the next section we will show that Eq.(4.4) can also be used to fit  
 571 the pull-off force data for some shear separation cases.

572

573 **4.2 Shear separation**

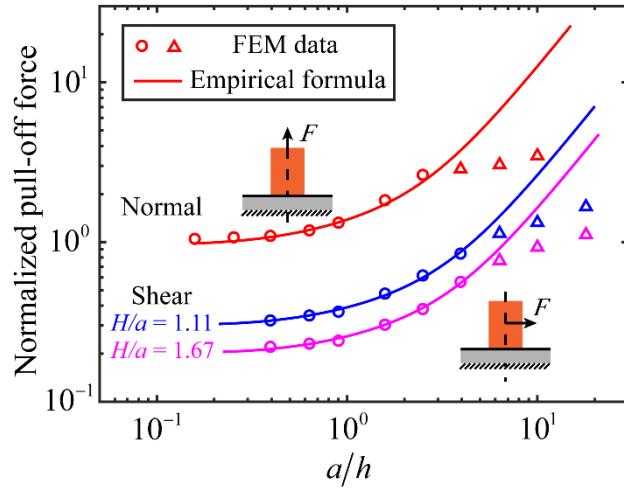
574 In this section, we consider shear separation ( $\alpha = 0$ ) and focus on computing the pull-  
 575 off force for different substrate thicknesses ( $a/h$ ) and loading positions ( $H/a$ ) and understanding  
 576 the corresponding delamination mechanisms.

577 **4.2.1 Cases with large  $H/a$**

578 We start with cases with large  $H/a$ . Specifically, Fig.9 plots the normalized pull-off  
 579 force  $\bar{F}_c$  defined in Eq.(4.2) versus  $a/h$  for  $H/a = 1.11$  and  $1.67$ . The normalized pull-off force  
 580 under normal separation is also plotted as a reference for comparison. Clearly the pull-off force

581 under shear separation is much lower than that under normal separation. In addition, shear  
 582 separation with a larger  $H/a$  results in a smaller pull-off force.

583



584

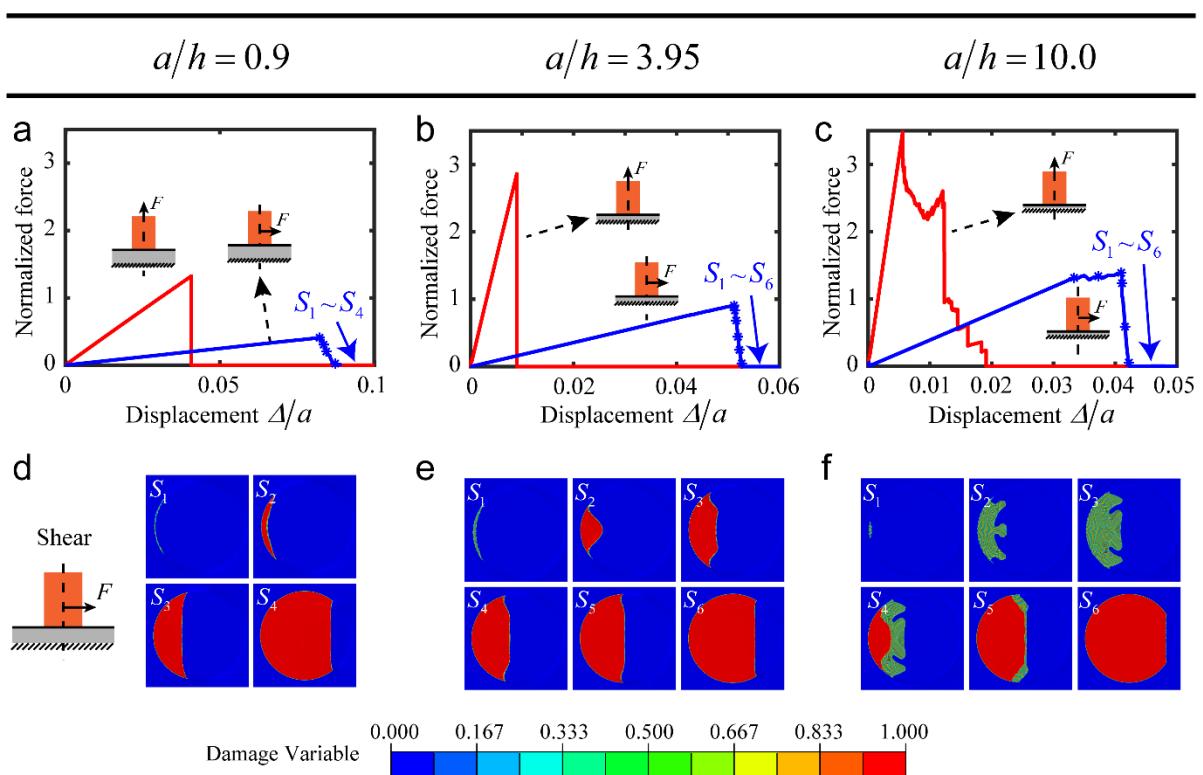
585 **Figure 9** Normalized pull-off force versus  $a/h$  under shear separation with  $H/a = 1.11$  and  $1.67$ .  
 586 The normal separation case is also included for comparison. The symbols (circles: Mode-I  
 587 crack propagation; triangles: interface cavitation) represent our FE results. The solid lines are  
 588 obtained by fitting Eq.(4.4) to the FE data with the corresponding coefficients given in Table  
 589 1.

590

591 To understand the mechanism underlying the lowered pull-off force, the evolution of  
 592 delamination for  $H/a = 1.11$  and three representative substrate thickness ( $a/h = 0.9, 3.95, 10$ ) are  
 593 illustrated in Fig.10 using the color maps of the interface damage variable  $D$ . For thick substrate  
 594 (e.g.,  $a/h = 0.9$ ), unlike normal separation where delamination initiates along the periphery of  
 595 the contact area (see Fig.8d), Fig.10d shows that delamination initiates at the left edge of the  
 596 contact area, followed by an unstable growth of delaminated region across the contact area  
 597 (assuming force control). Such localized delamination, caused by rotation of the rigid punch  
 598 driven by the rigid punch and the consequent stress concentration at the left edge, leads to the  
 599 lower pull-off force under shear separation. This mechanism also implies that increasing  $H/a$   
 600 can reduce the pull-off force by promoting rotation of the punch, which is consistent with our  
 601 observations in the simulation. When the substrate becomes very thin (e.g.,  $a/h = 10$ ), interface  
 602 cavitation can be observed in Fig.10f, but is biased towards the left edge due to the stress

603 concentration caused by punch rotation. In Fig.9 we use triangular and circular symbols to  
 604 denote the FE data for pull-off forces with and without interface cavitation, respectively.  
 605 Similar to normal separation, the triangular symbols are much lower than the trend extrapolated  
 606 from the circular symbols, indicating that interface cavitation can also reduce the pull-off force  
 607 under shear separation.

608 Since the shear separation with thick substrates is governed by local Mode-I interface  
 609 crack propagation, the pull-off force data for shear separation in Fig.9 follow the similar trend  
 610 as the pull-off force for normal separation. Indeed, we find that the shear pull-off force data  
 611 can be well captured by shifting the normal pull-off force curve in the log-log plot of Fig.9.  
 612 This enables a simple way of obtaining empirical pull-off force formulas for the shear  
 613 separation cases by rescaling  $\bar{F}_c$  and  $a/h$  based on the shifting factors. As a result, we find  
 614 that the pull-off force data under shear separation (circular symbols in Fig.9) can still be well  
 615 captured by Eq.(4.4), and the corresponding values of  $C_1$  to  $C_5$  are listed in Table 1 below.  
 616



617  
 618 **Figure 10** Delamination behavior under shear separation with  $H/a=1.11$  and different  $a/h$ . (a-  
 27

619 c) The force-displacement curves during delamination. The corresponding force-displacements  
 620 for normal separation is included for comparison. (d-f) Evolution of the interface damage  
 621 variable  $D$  (shown in the top view of the substrate surface) to quantity local interface  
 622 delamination: (d)  $a/h = 0.9$ , (e)  $a/h = 3.95$ , (f)  $a/h = 10.0$ .

623

624 **Table 1:** Coefficients for the empirical formula Eq.(4.4) of pull-off force under normal or shear  
 625 separation.

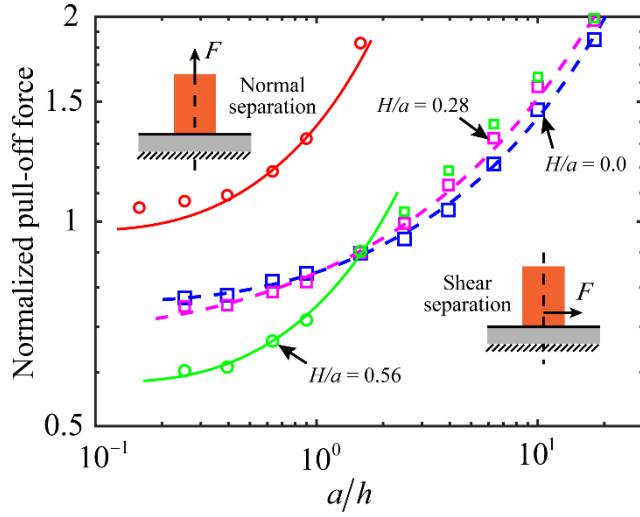
$H/a$	Valid for	$C_1$	$C_2$	$C_3$	$C_4$	$C_5$
<b>Normal separation</b>						
For any $H/a$	$a/h \leq 2.5$	$\frac{8}{3}\sqrt{2\pi}$	2.2002	0.2864	5.8186	0.7427
<b>Shear separation</b>						
1.67	$a/h \leq 3.95$	1.3973	1.5663	0.1033	4.1422	0.2680
1.11	$a/h \leq 3.95$	2.0892	1.6472	0.1202	4.3562	0.3117
0.56	$a/h \leq 1.58$	3.9992	1.6715	0.1256	4.4204	0.3256

626 4.2.2 Cases with small  $H/a$

627 Besides the two delamination modes discussed above (i.e., Mode-I crack propagation  
 628 and interface cavitation), a new delamination mode for shear separation emerges if  $H/a$  is  
 629 further decreased. To see that, we plot the normalized pull-off force versus  $a/h$  for three cases  
 630 of  $H/a$  ( $= 0.56, 0.28, 0$ ) in Fig.11. Specifically, when  $H/a = 0.56$ , the pull-off force follows the  
 631 empirical formula in Eq.(4.4) for substrates with large to moderate thickness ( $a/h \leq 1.58$ ). The  
 632 corresponding fitting parameters are listed in Table 1. However, for thin substrates ( $a/h \geq 2.5$ ),  
 633 the pull-off force data points deviate from the trend given by Eq.(4.4) (see Fig.11). To  
 634 understand the physical nature of this deviation, the evolution of interface delamination for  $H/a$   
 635  $= 0.56$  and three cases of  $a/h$  ( $= 0.9, 2.5, 10.0$ ) is shown in Fig.12a. Interestingly, when the  
 636 substrate is thick ( $a/h = 0.9$ ), we observe the same delamination process illustrated in Fig.10d,  
 637 i.e., delamination initiates at the left edge of the contact area due to rotation of the punch and  
 638 then spreads across the entire contact area. In contrast, when the substrate is thin ( $a/h = 2.5$  or

639 10), delamination initiates at both the top and bottom edges of the contact area. This is because  
 640 thin substrate is much less compliant along the normal direction, which suppresses the punch  
 641 rotation under the delamination force  $F$ . As a result, the contacting area is primarily subjected  
 642 to shear tractions, and thus delamination is mainly due to a Mode-II interface crack propagation.  
 643 We will refer to this mode as Mode-II crack propagation.

644



645 **Figure 11** Normalized pull-off force versus  $a/h$  under shear separation with  $H/a = 0, 0.28$  and  
 646  $0.56$ . The normal separation case is also included for comparison. The symbols (circles: Mode-  
 647 I crack propagation; squares: Mode-II crack propagation) represent our FE results. The solid  
 648 lines are obtained by fitting Eq.(4.4) to the FE data with the corresponding coefficients given  
 649 in Table 1. The dashed lines are given by Eq.(4.5) and Eq.(4.6).

650

651 When  $H/a$  is further reduced (i.e., 0.28 and 0 in Fig.11), the Mode-II crack propagation  
 652 mode prevails and the pull-off force becomes insensitive to  $H/a$ . In particular, Fig.12b shows  
 653 that when  $H/a = 0$ , delamination always initiates at the top and bottom edges even for thick  
 654 substrate ( $a/h = 0.9$ ). In this mode, the empirical formula in Eq.(4.4) can no longer capture the  
 655 FE data of pull-off forces. Instead, we find that the following expression by modifying the form  
 656 of Eq.(4.4) and fitting the FE data for  $H/a = 0$  and 0.28 in Fig.11, respectively:

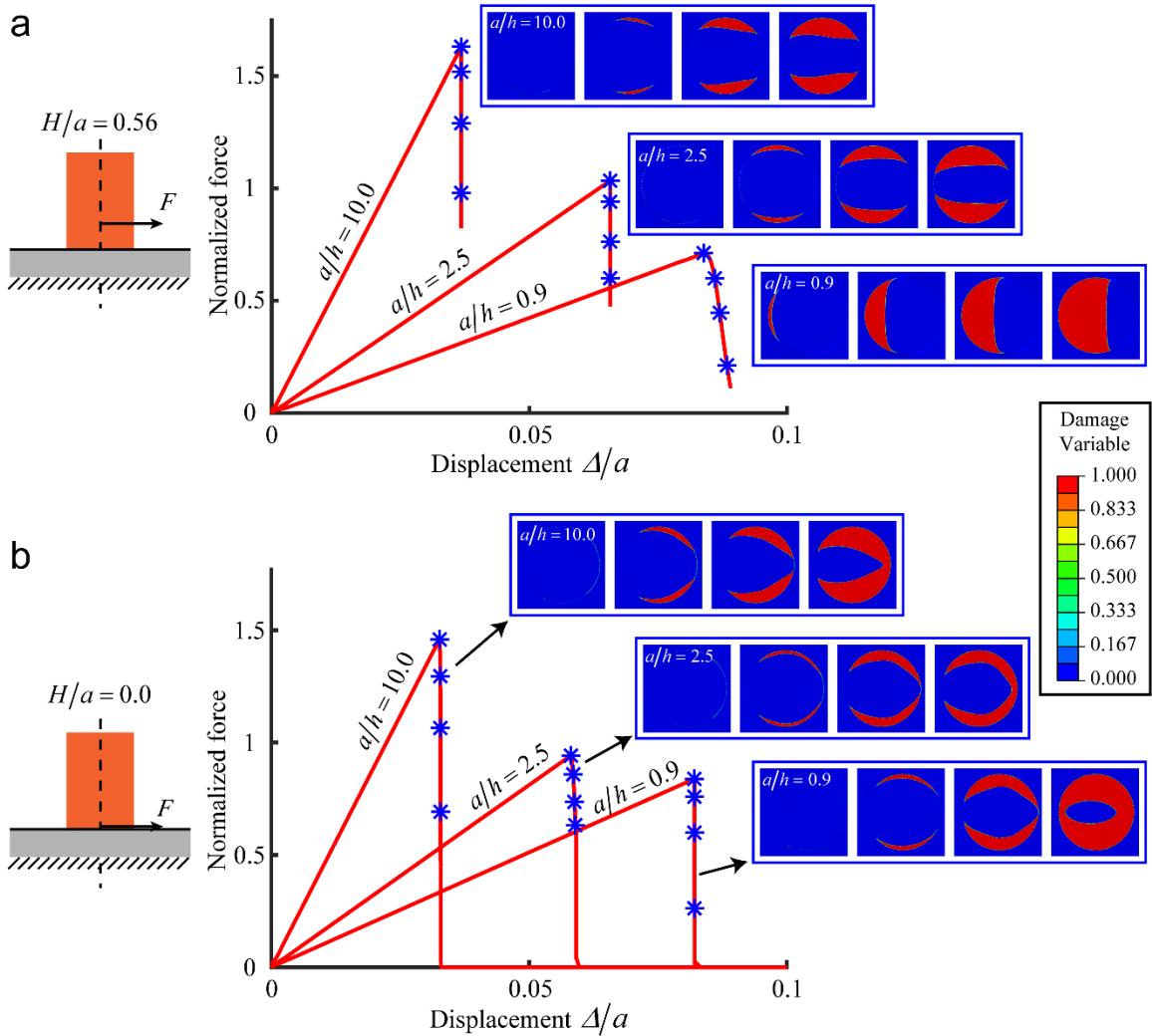
657

$$F_c = 5.2537 (W_{ad} E a^3)^{1/2} \left[ \frac{0.75 + 0.8519 (a/h)^{0.64} + 0.0166 (a/h)^{1.93}}{\left( 0.75 + 2.253 (a/h)^{0.64} + 0.0431 (a/h)^{1.93} \right)^{1/2}} \right], \quad H/a = 0.0 \quad (4.5)$$

658

$$F_c = 4.7616 \left( W_{ad} E a^3 \right)^{1/2} \left[ \frac{0.75 + 1.3492 (a/h)^{0.55} + 0.066 (a/h)^{1.65}}{\left( 0.75 + 3.5682 (a/h)^{0.55} + 0.1713 (a/h)^{1.65} \right)^{1/2}} \right], H/a = 0.28. \quad (4.6)$$

659



660

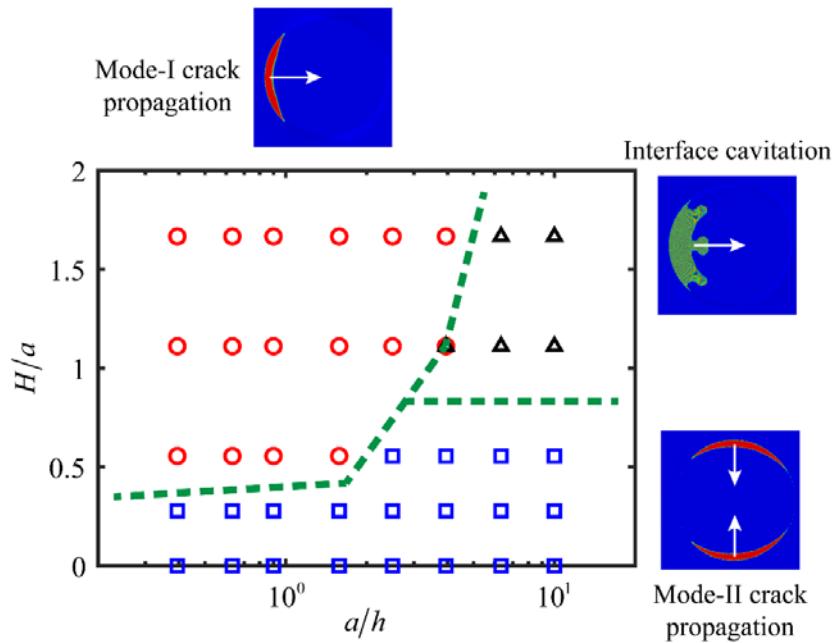
661 **Figure 12** The force-displacement curves and the corresponding evolutions of interface  
 662 delamination (represented by the color map of damage variable  $D$ ) for small values of  $H/a$ : (a)  
 663  $H/a=0.56$  and (b)  $H/a=0$ . For each  $H/a$ , three cases with different substrate thicknesses are  
 664 studied:  $a/h = 0.9, 2.5$  and  $10.0$ .

665

#### 666 4.2.3 Discussions

667 We have identified three different delamination modes under shear separation: i) Mode-  
 668 I crack propagation, ii) Mode-II crack propagation, and iii) interface cavitation. Which mode

669 would occur depends on the stress state in the elastic substrate developed during delamination,  
 670 which is governed by essentially two parameters:  $H/a$  representing height of the applied shear  
 671 force and  $a/h$  representing the substrate thickness. Using the FE data, we plot a phase diagram  
 672 of the delamination modes in Fig.13. Briefly, either the Mode-I crack propagation mode  
 673 (circular symbols) or interface cavitation mode (triangular symbols) would occur at large  $H/a$   
 674 where rotation of the rigid punch results in a region with concentrated normal traction near the  
 675 left edge of the contact area. Delamination tends to occur through interface cavitation for very  
 676 thin substrates (i.e., large  $a/h$ ), and through Mode-I crack propagation for thicker substrates.  
 677 On the other hand, the Mode-II crack propagation (square symbols) occurs when the punch  
 678 rotation is suppressed by the small  $H/a$ .



679

680 **Figure 13** Phase diagram of the three delamination modes with respect to  $H/a$  and  $a/h$ .

681

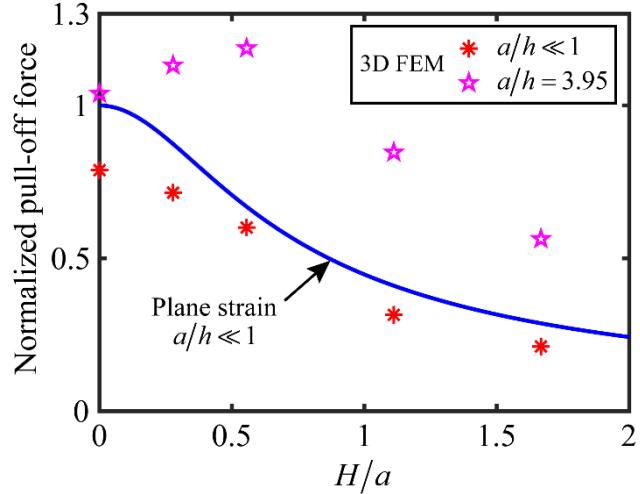
682 The effect of punch rotation on the pull-off force observed in the 3D FE model is  
 683 qualitatively similar to that in the 2D plane strain model. To illustrate this point, recall the  
 684 analytical solution of normalized pull-off force for the 2D model given in Eq.(2.27). By setting  
 685  $\alpha=0$ , we obtain

686

$$\bar{F}_c = \frac{1}{\sqrt{4(H/a)^2 + 1}} \quad . \quad (4.7)$$

687 Note that this normalized pull-off force of the 2D model is equivalent to that of the 3D model  
 688 (see Eq.(4.2)): both definitions use the corresponding normal pull-off force for infinitely thick  
 689 substrates as the reference. As shown in Fig.14, Eq.(4.7) predicts that  $\bar{F}_c$  scales with  $(H/a)^{-1}$   
 690 when  $H/a \gg 1$ . This trend is consistent with the 3D case with thick substrate ( $a/h \ll 1$ ) which  
 691 is obtained by extrapolating the empirical formulas in Eqs.(4.4)-(4.6) to the limit of  $a/h \rightarrow 0$ ,  
 692 and is illustrated in Fig.14 using asterisk symbols. Indeed, the quantitative agreement between  
 693 the 2D solution and 3D data for thick substrate ( $a/h \ll 1$ ) is remarkable. This suggests that  
 694 Eq.(4.7) can be used as a quantitative guide for estimating the effect of the height  $H$  on pull-  
 695 off force for thick substrates. Interestingly, the scaling relation that the pull-off force  $F_c \sim$   
 696  $(H/a)^{-1}$  when  $H/a \gg 1$  was also observed in the experimental data of Chaudhury and Kim<sup>20</sup>,  
 697 even though our 2D model assumes an elastic half space ( $a/h \ll 1$ ) while the experimental data<sup>20</sup>  
 698 were for thin substrates ( $a/h \gg 1$ ).

699 The FE data for a case of thin substrate ( $a/h = 3.95$ ) are also plotted in Fig.14 for  
 700 comparison. Interestingly, a non-monotonic dependence of the pull-off force on  $H/a$  is observed,  
 701 which is due to the transition of delamination modes as  $H/a$  is increased. As shown in Fig.13,  
 702 when  $a/h = 3.95$  and  $H/a < \sim 0.56$ , the dominating delamination mode is Mode-II crack  
 703 propagation. With this mode, increasing  $H/a$  can cause slight rotation of the punch which may  
 704 reduce the energy release rate associated with the Mode-II interface crack and hence lead to an  
 705 increase in pull-off force. However, when  $H/a > \sim 0.56$ , the dominating delamination mode  
 706 becomes Mode-I crack propagation where increasing  $H/a$  can reduce the pull-off force by  
 707 promoting punch rotation and thus enhancing local Mode-I energy release rate at the left edge  
 708 of the contact area.



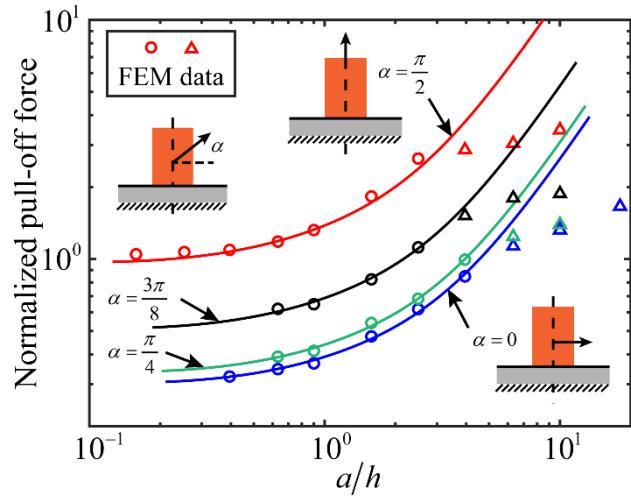
709

710 **Figure 14** Normalized pull-off force versus  $H/a$  given by the 2D plane strain model (solid line)  
 711 and 3D model for  $a/h \ll 1$  (asterisks) and  $a/h = 3.95$  (stars).

712

713 **4.3 Combined shear and normal forces: effect of loading angle**

714 In this section, we consider the scenario where the punch is subjected to combined  
 715 shear and normal forces, with the resultant force pointing along an angle  $\alpha$  above the horizontal  
 716 direction (see Fig.5a). Our focus is on how the loading angle  $\alpha$  affects the pull-off force, and  
 717 therefore we fix  $H/a = 1.11$  but vary the substrate thickness (i.e.,  $a/h$ ). The FE data of pull-off  
 718 forces are shown in Fig.15, which shows that the pull-off force increases as the loading angle  
 719 increases from 0 to  $\pi/2$ . In this set of simulations, we only observe two delamination modes:  
 720 Mode-I crack propagation (circular symbols) and interface cavitation (triangular symbols),  
 721 similar to the case of shear separation with large  $H/a$  (see Fig.9 and 10). In addition, the pull-  
 722 off force resulting from the Mode-I crack propagation mode can also be well fitted by Eq.(4.4)  
 723 through shifting the normal pull-off force curve, and the coefficients  $C_1$  to  $C_5$  for different  
 724 angles  $\alpha$  are summarized in Table 2.



725

726 **Figure 15** Normalized pull-off force versus  $a/h$  with different loading angles  $\alpha$  and  $H/a = 1.11$ .  
727 The symbols (circles: Mode-I crack propagation; triangles: interface cavitation) represent our  
728 FE results. The solid lines are obtained by fitting Eq.(4.4) to the FE data with the corresponding  
729 coefficients given in Table 2.

730

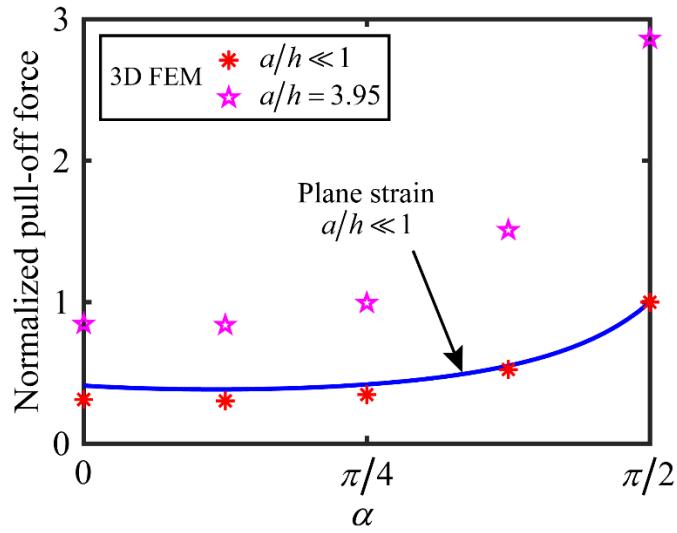
731 **Table 2** Coefficients for the empirical formula Eq.(4.4) of pull-off force under combined  
732 normal and shear forces with  $H/a = 1.11$  and different angles  $\alpha$ .

$\alpha$	Valid for	$C_1$	$C_2$	$C_3$	$C_4$	$C_5$
0	$a/h < 3.95$	2.0892	1.6472	0.1202	4.3562	0.3117
$\pi/8$	$a/h < 3.95$	2.0224	1.9084	0.1869	5.0469	0.4847
$\pi/4$	$a/h < 3.95$	2.3198	1.7146	0.1355	4.5344	0.3515
$3\pi/8$	$a/h < 2.5$	3.5041	1.8739	0.1770	4.9558	0.4589
$\pi/2$	$a/h < 2.5$	$\frac{8}{3}\sqrt{2\pi}$	2.2002	0.2864	5.8186	0.7427

733

734 Interestingly, the FE results suggest that the pull-off force is insensitive to the loading  
735 angle when  $\alpha$  is below  $\pi/4$ . When  $\alpha$  exceeds  $\pi/4$ , the pull-off force rapidly increases with  $\alpha$   
736 and approaches the limit of normal separation. To illustrate this behavior, in Fig.16 we plot the  
737 dependence of normalized pull-off force  $\bar{F}_c$  on the loading angle  $\alpha$ . For the 3D model, we  
738 include two cases: i) thick substrate ( $a/h \ll 1$ ) by extrapolating Eq.(4.4) with the coefficients

739 given in Table 2, and ii) a case representing thin substrate ( $a/h = 3.95$ ). The qualitatively trend  
 740 of how  $\bar{F}_c$  on the loading angle  $\alpha$  for both thick and thin substrates agrees with that of the 2D  
 741 model predicted by the analytical solution Eq.(2.27). Again, this agreement shows that Eq.(2.27)  
 742 derived from the 2D model can be used for estimating the effect of loading angle in the 3D  
 743 case with cylindrical punch.



744  
 745 **Figure 16** Normalized pull-off force versus the loading angle for  $H/a = 1.11$  given by the 2D  
 746 plane strain model (solid line) and 3D model for  $a/h \ll 1$  (asterisks) and  $a/h = 3.95$  (stars).

747

748 **5. Summary and Conclusions**

749 We presented a theoretical and computational study on the delamination of a rigid  
 750 punch from an elastic substrate under normal and shear forces. We first studied a 2D plane  
 751 strain model to gain theoretical insights and then developed a 3D FE model to simulate the  
 752 complex 3D mechanics involved in the delamination process. From the 2D model, we derived  
 753 an analytical solution for the pull-off force using a fracture mechanics approach. This solution,  
 754 although based on the assumption of an elastic half-space substrate, was found to capture the  
 755 quantitative trend of how pull-off force depend on the loading position  $H$  and loading angle  $\alpha$   
 756 exhibited in the 3D FE results with thick substrates. Both the 2D analytical model and 3D FE  
 757 model show that given the same properties of substrate (i.e.,  $E$  and  $h$ ) and interface (i.e.,  $W_{ad}$   
 758 and  $a$ ), normal separation requires the largest pull-off force. The main physical mechanism

759 behind the lower pull-off force for shear or angled separation is the uneven normal traction due  
760 to the punch rotation, which causes delamination to initiate locally near an edge of the contact  
761 area as opposed to along the entire periphery during normal separation. For shear separation,  
762 we identified three delamination modes: Mode-I crack propagation, Mode-II crack propagation  
763 and interface cavitation. Empirical formulas were obtained for pull-off forces governed by the  
764 first two modes.

765 Our results have practical implications for the design of fouling release or anti-icing  
766 coatings. Since the delamination force in reality may include both normal and shear  
767 components, the empirical formulas in our work can provide more accurate estimate of the  
768 pull-off force, thereby facilitating the development of more efficient release methods. For  
769 example, it is advantageous to apply shear force during delamination which can greatly reduce  
770 the pull-off force. In addition, a common principle of the delamination modes with low pull-  
771 off force is to promote the initiation of local delamination, either through interface cavitation  
772 or the introduction of uneven tractions on the contacting interface. This principle is consistent  
773 with the approach of interface crack initiator recently exploited to improve the release  
774 performance of anti-icing coatings<sup>44</sup>.

775 There are several limitations in this study that could be addressed in future work. First,  
776 we assumed infinitesimal deformation and linear elasticity for the substrate to enable analytical  
777 solution for the 2D model and to reduce independent parameters for the 3D FE model. Although  
778 this is a relevant assumption for many coating applications, large deformation may occur for  
779 strong adhesion and soft substrate. In this case, nonlinearity associated with large deformation  
780 may further complicate the delamination mechanics<sup>45,46</sup>. This question remains to be answered.  
781 Second, we have not discussed in detail the pull-off force governed by the interface cavitation  
782 mode. The FE simulation of interface cavitation would require an extensive imperfection  
783 sensitivity study to validate the FE results, but this will be necessary if thin coatings are  
784 encountered (i.e.,  $a/h \gg 1$ ). Third, in practice the rigid object to be detached from the elastic  
785 substrate possesses more complex structure<sup>21,47</sup>. For example, barnacles, a common marine  
786 fouling organism, are known to have a hollow shell structure rather than a solid punch. The

787 substrate may not be uniform either: mechanical heterogeneities may be deliberately  
788 incorporated to promote local delamination on the interface. How these complex structures  
789 affect the delamination process, especially under shear or angled separation, remains an open  
790 question and requires further studies.

791

## 792 Acknowledgements

793 X.H.S. and H.A.W. acknowledge support from the Strategic Priority Research Program  
794 of the Chinese Academy of Sciences (XDB22040402) and the National Natural Science  
795 Foundation of China (11525211, 11872036). L.Y., M.E.R. and R.L. acknowledge National  
796 Science Foundation for funding this work through grant CMMI-1636203. This work used the  
797 Extreme Science and Engineering Discovery Environment (XSEDE) through allocation  
798 MSS160030, which is supported by National Science Foundation grant number ACI-1548562.

799

## 800 Appendix 1 Finite element model: effect of mesh size and cohesive parameter.

801 In our simulations, the cohesive zone models and mesh designs will significantly impact  
802 the FE results of pull-off forces. To validate our computational model, we perform sensitivity  
803 studies regarding the cohesive zone model and mesh size as detailed below.

804 Two representative traction-separation laws for the cohesive zone model tested in the  
805 simulation are shown in Fig.17b. In both cases, the following adhesion energy  $W_{ad}$  is adopted  
806 to ensure a small deformation of the substrate:

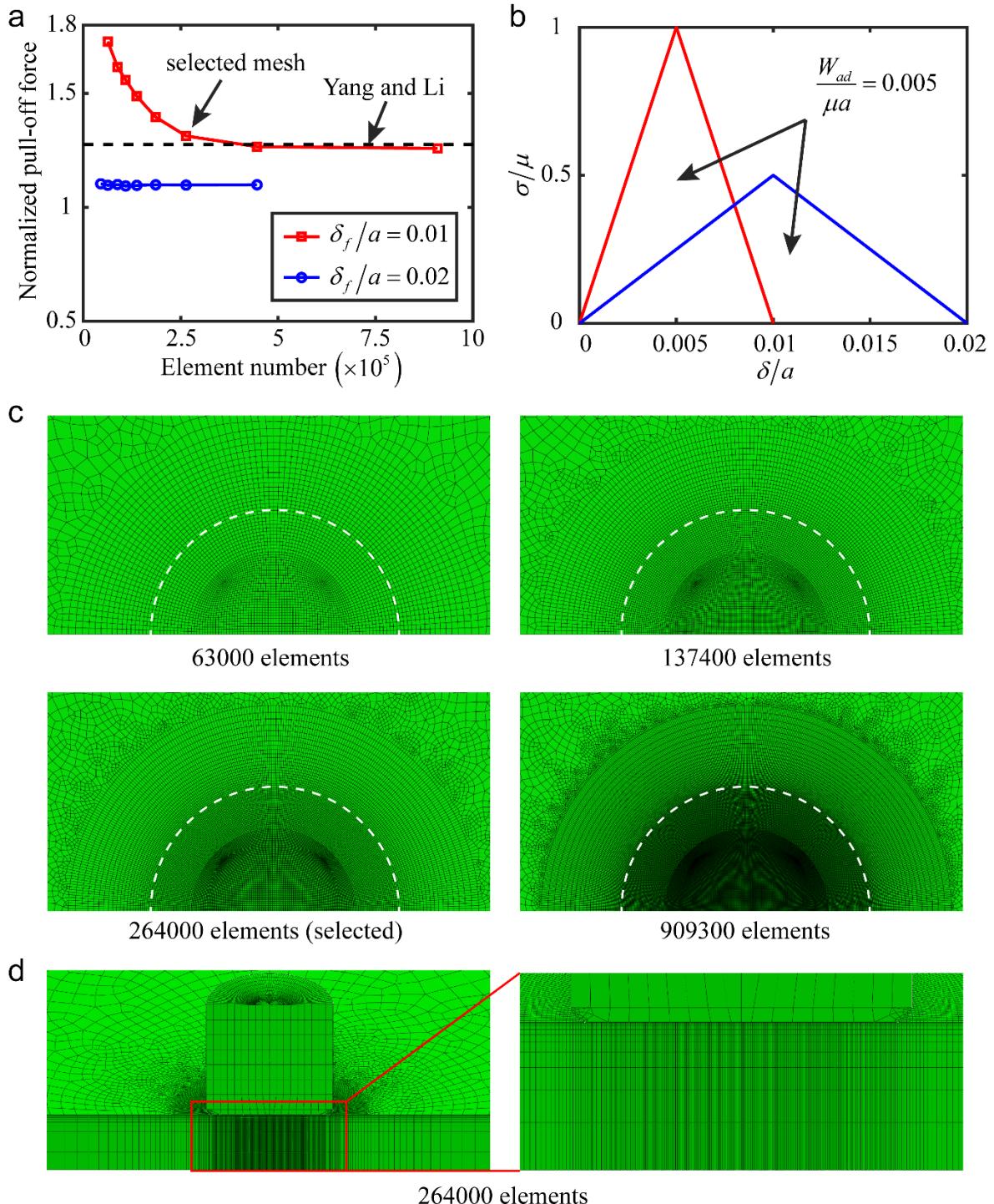
$$807 \frac{W_{ad}}{\mu a} = 0.005, \quad (A1.1)$$

808 where  $\mu=E/3$  denotes the shear modulus of the substrate. The maximum separation  $\delta_f$  values,  
809 which are much smaller than the contact radius  $a$ , i.e.  $\delta_f/a=0.01$  and  $0.02$ , are chosen such that  
810 the detailed shape of the traction-separation curve will have negligible effects on the interface  
811 adhesive behavior<sup>38</sup>. For each traction-separation law, the mesh convergence test is performed

812 to obtain a mesh-independent pull-off force, as shown in Fig.17a.

813 We focus on the normal separation case with a moderate substrate thickness ( $a/h=0.9$ )  
814 and use the numerical solution for the pull-off force by Yang and Li<sup>17</sup> shown in Fig.17a as a  
815 benchmark. The FE data for pull-off forces with two traction-separation laws are plotted against  
816 the element numbers, as shown in Fig.17a. Note that the forces are normalized to the normal  
817 pull-off force  $F_{c0}$  for an elastic half-space ( $a/h=0$ ) and the dashed line represents the solution  
818 of Yang and Li<sup>17</sup>. As can be seen in Fig.17a, the pull-off forces for both traction-separation laws  
819 get converged as the mesh is refined. However, the pull-off force for  $\delta_f/a = 0.01$  has achieved a  
820 better agreement with the solution by Yang and Li<sup>17</sup>. This is expected since an interface in linear  
821 fracture mechanics can be theoretically characterized by a traction-separation law in the Dirac-  
822 function form, implying that a smaller  $\delta_f/a$  can result in a more accurate simulation of the  
823 interface fracture behavior. Another implication by Fig.17a is that a smaller  $\delta_f/a$  will require a  
824 more refined mesh design to achieve the mesh-independent results. Interestingly, when the  
825 element size at the edge of the contact region is exactly equal to  $\delta_f$ , mesh convergence of the  
826 pull-off force is achieved. This means we cannot infinitely reduce  $\delta_f/a$  considering the  
827 computational costs for reliable results. As a result, to balance computational cost and  
828 numerical accuracy, we choose the interface parameter of  $\delta_f/a=0.01$  and the mesh design of  
829 264000 elements (see Fig.17d) to perform the delamination simulations in the present work.

830



831

832 **Figure 17** The mesh convergence tests with two cohesive zone models. (a) Normalized pull-  
 833 off force versus element numbers under normal separation for the case of  $a/h=0.9$  with two  
 834 traction-separation laws. The corresponding numerical result by Yang and Li<sup>17</sup>, shown in  
 835 dashed line, is used as a benchmark. (b) Two representative traction-separation curves with the  
 836 same adhesion energy ( $W_{ad}/(\mu a)=0.005$ ) but different  $\delta_f/a$  values (red: 0.01; blue: 0.02), where  
 837  $\mu=E/3$  denotes the substrate shear modulus. (c) Several examples of the mesh designs of the  
 838 contact surface used in the simulation. The white dashed line depicts the edge of contacting

839 region. (d) Selected mesh design of the lateral cross-section. The inset shows a magnified view  
840 of the mesh.

841 **References**

- 842 1. Schultz, M. P. Effects of coating roughness and biofouling on ship resistance and  
843 powering. *Biofouling* **23**, 331–341 (2007).
- 844 2. Piola, R. F., Dafforn, K. A. & Johnston, E. L. The influence of antifouling practices on  
845 marine invasions. *Biofouling* **25**, 633–644 (2009).
- 846 3. Evans, S. ., Birchenough, A. . & Brancato, M. . The TBT Ban: Out of the Frying Pan  
847 into the Fire? *Mar. Pollut. Bull.* **40**, 204–211 (2000).
- 848 4. Voulvoulis, N., Scrimshaw, M. D. & Lester, J. N. Alternative antifouling biocides.  
849 *Appl. Organomet. Chem.* **13**, 135–143 (1999).
- 850 5. Ellis, D. V & Agan Pattisina, L. Widespread neogastropod imposex: A biological  
851 indicator of global TBT contamination? *Mar. Pollut. Bull.* **21**, 248–253 (1990).
- 852 6. Lejars, M., Margaillan, A. & Bressy, C. Fouling Release Coatings: A Nontoxic  
853 Alternative to Biocidal Antifouling Coatings. *Chem. Rev.* **112**, 4347–4390 (2012).
- 854 7. Krishnan, S., Weinman, C. J. & Ober, C. K. Advances in polymers for anti-biofouling  
855 surfaces. *J. Mater. Chem.* **18**, 3405 (2008).
- 856 8. Salta, M. *et al.* Designing biomimetic antifouling surfaces. *Philos. Trans. A. Math.*  
857 *Phys. Eng. Sci.* **368**, 4729–54 (2010).
- 858 9. Grozea, C. M. & Walker, G. C. Approaches in designing non-toxic polymer surfaces  
859 to deter marine biofouling. *Soft Matter* **5**, 4088 (2009).
- 860 10. Brady, R. F. Clean hulls without poisons: Devising and testing nontoxic marine  
861 coatings. *J. Coatings Technol.* **72**, 45–56 (2000).
- 862 11. Zhuo, Y. *et al.* Enhancing the Mechanical Durability of Icephobic Surfaces by  
863 Introducing Autonomous Self-Healing Function. *ACS Appl. Mater. Interfaces* **10**,  
864 11972–11978 (2018).

865 12. Wang, C., Fuller, T., Zhang, W. & Wynne, K. J. Thickness Dependence of Ice  
866 Removal Stress for a Polydimethylsiloxane Nanocomposite: Sylgard 184. *Langmuir*  
867 **30**, 12819–12826 (2014).

868 13. Beemer, D. L., Wang, W. & Kota, A. K. Durable gels with ultra-low adhesion to ice. *J.  
869 Mater. Chem. A* **4**, 18253–18258 (2016).

870 14. Golovin, K. *et al.* Designing durable icephobic surfaces. *Sci. Adv.* **2**, e1501496–  
871 e1501496 (2016).

872 15. Kendall, K. The adhesion and surface energy of elastic solids. *J. Phys. D. Appl. Phys.*  
873 **4**, 320 (1971).

874 16. Lin, Y. Y., Hui, C. Y. & Conway, H. D. Detailed elastic analysis of the flat punch  
875 (tack) test for pressure-sensitive adhesives. *J. Polym. Sci. Part B Polym. Phys.* **38**,  
876 2769–2784 (2000).

877 17. Yang, F. & Li, J. C. M. Adhesion of a Rigid Punch to an Incompressible Elastic Film.  
878 (2001). doi:10.1021/LA010409H

879 18. Singer, I. L., Kohl, J. G. & Patterson, M. Mechanical aspects of silicone coatings for  
880 hard foulant control. *Biofouling* **16**, 301–309 (2000).

881 19. Brady, R. F. & Singer, I. L. Mechanical factors favoring release from fouling release  
882 coatings. *Biofouling* **15**, 73–81 (2000).

883 20. Chaudhury, M. K. & Kim, K. H. Shear-induced adhesive failure of a rigid slab in  
884 contact with a thin confined film. *Eur. Phys. J. E* **23**, 175–183 (2007).

885 21. Chung, J. Y. & Chaudhury, M. K. Soft and Hard Adhesion. *J. Adhes.* **81**, 1119–1145  
886 (2005).

887 22. Kohl, J. G. & Singer, I. L. Pull-off behavior of epoxy bonded to silicone duplex  
888 coatings. *Prog. Org. Coatings* **36**, 15–20 (1999).

889 23. Crosby, A. J., Shull, K. R., Lakrout, H. & Creton, C. Deformation and failure modes of

adhesively bonded elastic layers. *J. Appl. Phys.* **88**, 2956 (2000).

24. Webber, R. E., Shull, K. R., Roos, A. & Creton, C. Effects of geometric confinement on the adhesive debonding of soft elastic solids. *Phys. Rev. E - Stat. Physics, Plasmas, Fluids, Relat. Interdiscip. Top.* **68**, 11 (2003).

25. Lakrout, H., Sergot, P. & Creton, C. Direct observation of cavitation and fibrillation in a probe tack experiment on model acrylic pressure-sensitive-adhesives. *J. Adhes.* **69**, 307–359 (1999).

26. Creton, C. & Lakrout, H. Micromechanics of Flat-Probe Adhesion Tests of Soft Viscoelastic Polymer Films. *J. Polym. Sci. Part B Polym. Phys.* **38**, 965–979 (2000).

27. Chaudhury, M. K., Chakrabarti, A. & Ghatak, A. Adhesion-induced instabilities and pattern formation in thin films of elastomers and gels. *Eur. Phys. J. E* **38**, 82 (2015).

28. Schultz, M. P., Kavanagh, C. J. & Swain, G. W. Hydrodynamic forces on barnacles: Implications on detachment from fouling-release surfaces. *Biofouling* **13**, 323–335 (1999).

29. Swain, G. W., Griffith, J. R., Bultman, J. D. & Vincent, H. L. The use of barnacle adhesion measurements for the field evaluation of non-toxic foul release surfaces. *Biofouling* **6**, 105–114 (1992).

30. Swain, G. W. & Schultz, M. P. The testing and evaluation of non-toxic antifouling coatings. *Biofouling* **10**, 187–197 (1996).

31. Shull, K. R. & Crosby, A. J. Axisymmetric Adhesion Tests of Pressure Sensitive Adhesives. *J. Eng. Mater. Technol.* **119**, 211 (1997).

32. Ganghoffer, J. F. & Gent, A. N. Adhesion of a Rigid Punch to a Thin Elastic Layer. *J. Adhes.* **48**, 75–84 (1995).

33. Hensel, R., McMeeking, R. M. & Kossa, A. Adhesion of a rigid punch to a confined elastic layer revisited. *J. Adhes.* 1–20 (2017). doi:10.1080/00218464.2017.1381603

915 34. Beemer, D. L., Wang, W. & Kota, A. K. Durable gels with ultra-low adhesion to ice. *J.  
916 Mater. Chem. A* **4**, 18253–18258 (2016).

917 35. Johnson, K. L. *Contact Mechanics*. (Cambridge University Press, 1985).  
918 doi:10.1017/CBO9781139171731

919 36. Adams, G. G. Frictional slip of a rigid punch on an elastic half-plane. *Proc. R. Soc. A  
920 Math. Phys. Eng. Sci.* **472**, 20160352 (2016).

921 37. Rice, J. R. Elastic Fracture Mechanics Concepts for Interfacial Cracks. *J. Appl. Mech.*  
922 **55**, 98 (1988).

923 38. Hui, C. Y., Ruina, A., Long, R. & Jagota, A. Cohesive Zone Models and Fracture. *J.  
924 Adhes.* **87**, 1–52 (2011).

925 39. Shull, K. R., Flanigan, C. M. & Crosby, A. J. Fingering instabilities of confined elastic  
926 layers in tension. *Phys. Rev. Lett.* **84**, 3057–3060 (2000).

927 40. Yang, F. Indentation of an incompressible elastic film. *Mech. Mater.* **30**, 275–286  
928 (1998).

929 41. Chadwick, R. S. Axisymmetric Indentation of a Thin Incompressible Elastic Layer.  
930 *SIAM J. Appl. Math.* **62**, 1520–1530 (2002).

931 42. Long, R., Hui, C.-Y., Kim, S. & Sitti, M. Modeling the soft backing layer thickness  
932 effect on adhesion of elastic microfiber arrays. *J. Appl. Phys.* **104**, 044301 (2008).

933 43. Chung, J. Y., Kim, K. H., Chaudhury, M. K., Sarkar, J. & Sharma, A. Confinement-  
934 induced instability and adhesive failure between dissimilar thin elastic films. *Eur.  
935 Phys. J. E* **20**, 47–53 (2006).

936 44. He, Z., Xiao, S., Gao, H., He, J. & Zhang, Z. Multiscale crack initiator promoted  
937 super-low ice adhesion surfaces. *Soft Matter* **13**, 6562–6568 (2017).

938 45. Lin, S. *et al.* Fringe instability in constrained soft elastic layers. *Soft Matter* **12**, 8899–  
939 8906 (2016).

940 46. Lin, S., Mao, Y., Radovitzky, R. & Zhao, X. Instabilities in confined elastic layers  
941 under tension: Fringe, fingering and cavitation. *J. Mech. Phys. Solids* **106**, 229–256  
942 (2017).

943 47. Hui, C.-Y., Long, R., Wahl, K. J. & Everett, R. K. Barnacles resist removal by crack  
944 trapping. *J. R. Soc. Interface* **8**, 868–79 (2011).

945