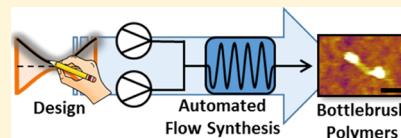


Engineering of Molecular Geometry in Bottlebrush Polymers

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Supporting Information

ABSTRACT: Bottlebrush polymers are large cylindrical macromolecules, where a molecular “width” emerges from a large number of side chains densely attached to a central backbone. The lengths of these grafted side chains exert indirect control over molecular flexibility, material processing, and molecular assembly behavior. Sequencing of the side-chain length is a promising route to further modify molecular geometry; however, tedious laboratory synthesis has imposed practical limits on this tunability. Here, we develop a methodology to overcome this limitation, leveraging automated synthesis and computer simulations to engineer bottlebrush polymers with three-dimensional molecular geometries. The automated flow synthesis platform combines fluid mechanics, reactor engineering, and living polymerization principles to gain precise synthetic control over the polymer architecture. Bottlebrush polymers with hourglass, football, bowtie, and sphere architecture profiles are synthesized with high molecular weights (up to 10^6 g mol⁻¹, ~150 nm) and narrow dispersities ($D < 1.1$). Atomic force microscopy and viscometry are used to illustrate the difference in the architecture of the polymers, providing results that match simulation predictions. This agreement enables the development of an inverse design protocol, where Monte Carlo simulations are used to correlate the molecular geometry for synthesis. This scalable synthetic strategy enables the production of *designer macromolecules* with *any* axisymmetric shape.



INTRODUCTION

The precise control over macromolecular chemical composition and shape is a hallmark of biological polymers. The hierarchical nature of biological structures gives rise to a wide variety of material properties and functionalities despite being built from a relatively small number of chemical building blocks (e.g., amino acids, base pairs, etc.).^{1–4} The branched architectures of graft proteins, such as lubricin,⁵ aggrecan,⁶ and mucin,⁷ play a major role in enabling their complex functionality. Mucins, for example, rely on their branched molecular architecture (layers of tethered brush- and gel-like mucins) to enable osmotic moduli gradients which aid in simultaneous mucus clearance and small molecule barrier properties.^{8,9} We aspire to emulate biological control over macromolecular properties by enabling the molecular design of polymer architectures (repeat unit connectivity) and polymer three-dimensional (3D) geometries (equilibrium conformation in solution).

Automated synthesis is a paradigm shift in molecule production as it has significantly accelerated and simplified the access to a wide array of molecules with biology-like complexity.^{10–13} This technological revolution is enabled by the development of laboratory-scale flow chemistry techniques, which are now being merged with the precision of living polymerizations. The primary structure of synthetic linear polymers (compositions, monomer sequences, molecular weights, and molecular weight distributions) can be precisely tuned *on-the-fly* using automated synthesis protocols.^{14–16} However, the systematic control of the 3D *molecular geometry* of polymers remains to be addressed as it requires control of the polymer *architecture* (i.e., the length and connectivity

between monomers) and predictive theoretical or computational models to link the architecture to the polymer *conformation* or *geometry* (i.e., the equilibrium spatial arrangement of these same monomers).

In this paper, we leverage the synergy between automated synthesis and computational simulations to create a methodology enabling the design and synthesis of polymers with precise macromolecular geometries (Figure 1). Our material platform is a class of highly branched polymers, known as bottlebrush polymers. These polymers adopt an elongated conformation and reduction in flexibility because of the steric repulsion among the side chains densely grafted to a molecular backbone.^{17–30} Generally, bottlebrush polymers possess a cylindrical molecular geometry, with the side chains giving rise to a “width” dimension and the backbone size translating to a “length”. A number of applications have benefited from the “semiflexible” rigid rod character of bottlebrush polymers: ultrasoft elastomers^{17,31–36} enabled by low cross-linking density and diminished conformational degrees of freedom and self-assembled structures (photonic crystals,^{35,37–49} drug delivery vehicles,^{50–63} etc.) with rapid assembly rates because of the suppression of molecular entanglements.⁶⁴ These polymers are attractive as a platform for *designer macromolecules* because the bottlebrush width can be indirectly controlled via synthesis through variation of the side-chain length. Recent efforts have expanded the scope of accessible architectures beyond a simple cylindrical geometry; however, they have

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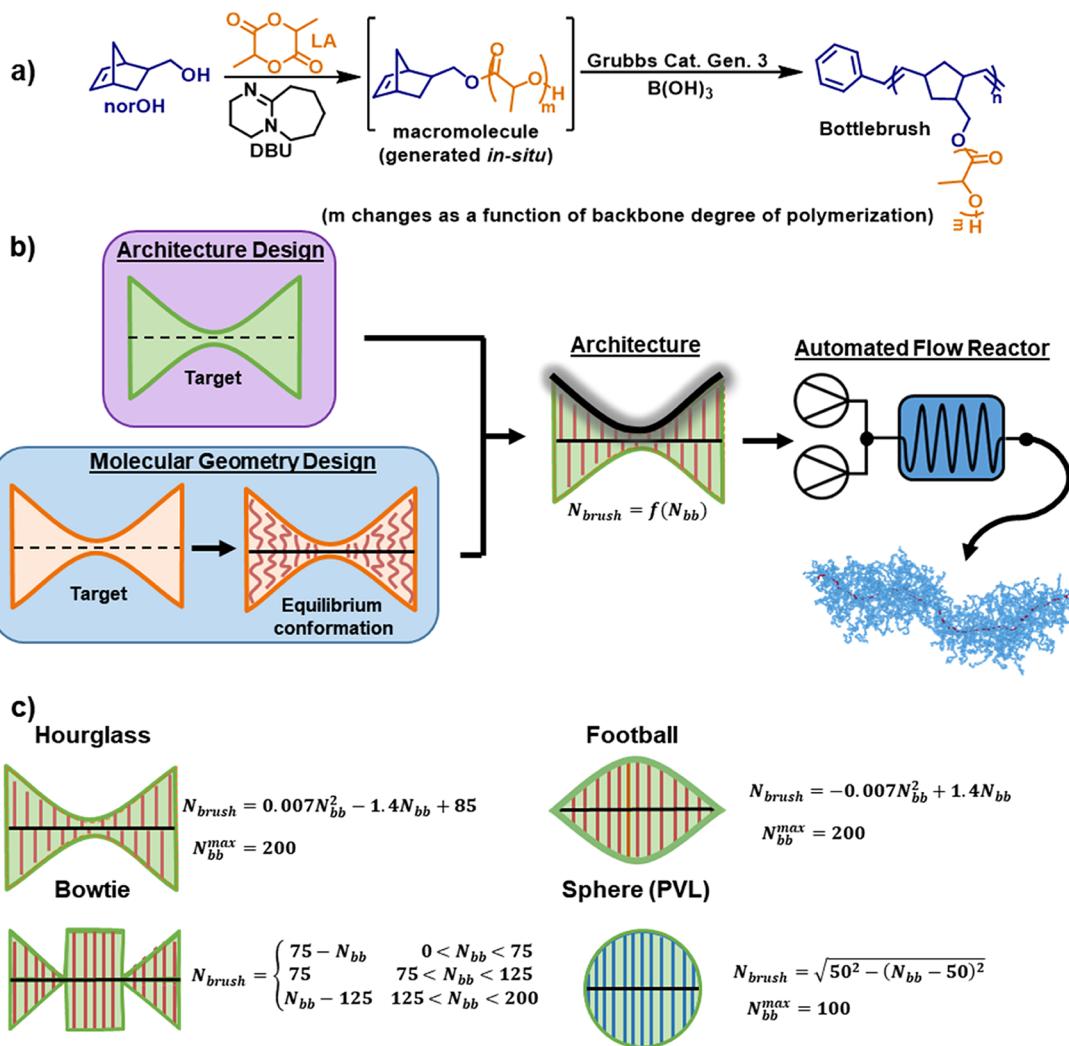


Figure 1. Overview of chemistry, design process, and design architectures. (a) Graft through synthesis of architecture-controlled bottlebrush polymers utilizing ROP of LA to produce macromonomers which are subsequently polymerized by third-generation Grubbs catalysts via ROMP. (b) Design workflow for the synthesis of bottlebrush. The upper route implements the targeted design directly to the architecture, whereas the bottom route implements an inverse design protocol to determine the architecture needed to produce a targeted molecular geometry. (c) Targeted bottlebrush architectures with the corresponding mathematical function synthesized in this manuscript.

suffered from tedious multistep synthesis or limited architecture control.^{65–74} New advances are required to overcome these practical synthetic limitations, including our prior work which is limited to monotonic changes in branch length. Moreover, these methods have dealt solely with architecture control and have not translated to the equilibrium space-filling polymer conformation, that is, molecular geometry. Addressing this challenge requires advances in polymer modeling; classical polymer theories on linear and branched architectures^{75–79} and more recent efforts considering bottlebrush architectures,^{26–28,80–83} focus on how molecular size and shape emerge from monomer connectivity. An inverse relationship for how monomers should be connected to yield a desired molecular size and shape is instead required to *design* shape-defined bottlebrush molecules.

The automated synthesis platform reported here gives access to molecules with *any* axisymmetric geometry from commercially available reagents. The flexibility of this approach is illustrated by synthesizing hourglass, football, spherical, and bowtie architectures using two different polymer brushes (polylactic acid and polyvalerolactone) (Figure 1). The

precision of the synthesis is illustrated through kinetics, atomic force microscopy (AFM), and viscometry experiments. The agreement between simulation predictions and experiments enables an inverse design protocol where molecular geometry is the target. The simplicity and high productivity of this strategy will accelerate the access to polymers with 3D *molecular structures* and provide a unique platform to emulate the structure function relationships observed in biological polymers.

RESULTS AND DISCUSSION

The first section outlines the methodology, chemistry, and automated flow reactor development. This is followed by the implementation and validation of the automated flow system to produce macromonomers. Subsequently, the synthesis of architecture-controlled bottlebrush polymers was performed and validated by imaging and viscometry measurements in conjunction with simulations. The final section leverages a Monte Carlo simulation to implement an inverse design protocol to calculate the architecture necessary to produce any specific axisymmetric molecular geometry.

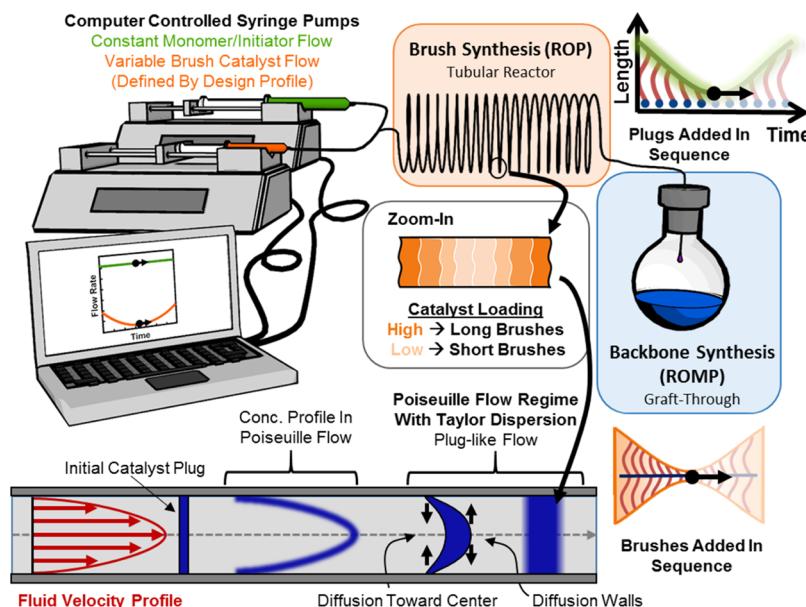


Figure 2. Automated flow reactor with bottlebrush synthesis and fluid mechanical elements. The setup consists of two computer-controlled syringe pumps connected to a long tubular reactor feeding into a collection pot. The green syringe contains the monomer and the initiator for the macromonomer synthesis and will flow at a constant rate. The orange syringe contains the catalyst and will flow at a variable rate defined by the design. The two solutions mix and enter the tubular reactor where the brush synthesis occurs. A zoom-in of the tubular reactor is present in the center of the figure detailing how each plug of fluid contains a different amount of catalyst. The higher the catalyst loading, the longer the brushes will be, and vice versa for low catalyst loading. The bottom of the figure provides a fluid mechanical description of the velocity profile (red), the initial catalyst plug, Poiseuille concentration profile without diffusion, and Poiseuille concentration profile with diffusion, where the blue color represents a single catalyst plug. The top right corner depicts the change in brush length as a function of time for an hourglass synthesis. The collection flask is where the brushes are added to a growing backbone via graft-through polymerization completing the synthesis of the architecture-controlled bottlebrush polymers.

Automated Flow Reactor Design. Bottlebrush polymers are produced in a two-step process where macromonomers are synthesized and then polymerized (graft-through polymerization).^{18,38,57,67,80–87} We hypothesized that the bottlebrush polymers with any axisymmetric architecture can be accessed by synthesizing macromonomers of tunable length in a tubular flow reactor and slowly feeding them into a second batch reactor where they are polymerized in the desired sequence (Figure 2). This methodology relies on two polymerizations, macromonomer synthesis and backbone synthesis, being chemically compatible and orthogonal. An additional requirement is the nearly instantaneous incorporation of the slowly added macromonomers into the bottlebrush structure to avoid scrambling of macromonomers. We have previously identified the cascade ring-opening polymerization (ROP) of cyclic esters for the synthesis of the brushes and ring opening metathesis polymerization (ROMP) of norbornene-terminated macromonomer initiated by a ruthenium-based catalyst (Grubbs' third-generation catalyst, G3) for the backbone synthesis to satisfy all synthetic requirements.^{67,88–91}

The synthesis of bottlebrush polymers with any axisymmetric architecture relies on the ability to precisely synthesize macromonomers of any length in a designed sequence. Per our hypothesis, we postulated that this control could be achieved by operating a computer-programmed flow reactor, where short plugs of fluid would form with different brush lengths because of different polymerization rates (Figure 2: zoom in). Turbulent flow regime (high Reynolds numbers) is the most common approach for producing plug flow; however, this is often impractical for laboratory scale as it requires the usage of large amounts of chemicals.^{16,92} In contrast, laminar flow

regime (low Reynolds numbers) can be easily implemented; however, this flow regime produces a parabolic velocity profile within the tube which causes a heterogeneity in residence time.⁹² Nevertheless, when slow flow rates, narrow tube diameters, and long tube lengths are implemented, it is possible to enter a Poiseuille flow regime where Taylor dispersion leads to plugs.^{93–101} This regime occurs when radial diffusion becomes significant over the reactor length (Figure 2: bottom).

To confirm that the automated flow reactor is capable of varying the macromonomer length *on-the-fly*, we performed the catalytic ROP of lactide (LA) and δ -valerolactone (VL) in the flow reactor without performing the subsequent graft-through polymerization of the backbone.^{18,38,67,80–87} The reactor setup consists of two computer-controlled syringe pumps: one syringe pump flowing the monomer/initiator solution and the other flowing the catalyst solution into a long and narrow tubular reactor (7.62 m with a 0.508 mm inside diameter; see *Supporting Information Section 2* for more details). The use of computer-controlled syringe pumps enables the automation and is a necessity to achieve the precise control over the polymer architecture. The degree of polymerization, which we correlate to the polymerization rate, can be controlled in the tubular reactor by varying the catalyst flow rate while maintaining a constant monomer/initiator flow rate. This designates the catalyst flow rate as the defining parameter of our process, where plugs containing high catalyst concentration will result in the formation of long macromonomers, whereas plugs with low catalyst concentration will yield short macromonomers.

The synthesis of any specific architecture requires a relationship between the targeted architecture profile (design parameter) and the catalyst flow rate (process parameter). Two independent approaches to correlate the catalyst flow rate with the macromonomer length were developed, as each method has its own merits. Method 1 implements the rate law for monomer consumption. A derivation using a general rate law that applies to most well-controlled polymerizations (eq 1),^{102,103} tubular reactor design equations,^{104,105} and the targeted architecture profile “*f*” (design profile) was performed to generate eq 2 (Supporting Information Section 4). The catalyst flow rate ($Q_{\text{syn}2}$, process parameter) can be calculated with eq 2 using a root-finding algorithm (a MATLAB code containing the secant method is provided in Supporting Information Section 12), and rate orders (*a*, *b*, *c*) can be determined from batch kinetics.

$$\frac{d[M]}{dt} = -k[I]^a[\text{cat}]^b[M]^c \quad (1)$$

$$\begin{aligned} \frac{Q_{\text{syn}2}^b}{(Q_{\text{syn}1} + Q_{\text{syn}2})^{a+b+c}} \\ = - \frac{F_{M,o}^{1-c}}{(c-1)(\pi R^2)LkF_I^a[\text{cat}_{\text{syn}2}]^b} \\ \left(1 - \left(1 - \frac{f\left(\frac{[I]_{\text{syn}1}Q_{\text{syn}1}^f}{n_{\text{bb,cat}}}\right)^{1-c}}{N_{\text{brush}}^{\text{synmax}}} \right) \right) \end{aligned} \quad (2)$$

To demonstrate the ability of method 1 to control the polymer length, three different architecture profiles (hourglass, football, and bowtie) were targeted by implementing the ROP of LA catalyzed by 8-diazabicyclo[5.4.0]undec-7-ene (DBU). In tetrahydrofuran (THF), the ROP of LA produces the reaction orders of *a* = 1, *b* = 1, and *c* = 1.8 (see Supporting Information Section 3, experimental details). For our experiments, a monomer/initiator solution ($[LA] = 0.92 \text{ M}$, $[LA]/[{\text{norOH}}] = 70$), a catalyst solution ($[DBU] = 0.3 \text{ M}$), a monomer/initiator flow rate of $0.125 \text{ mL min}^{-1}$, and a variable catalyst flow rate were used. Drops of the ROP reaction mixture were collected at the exit of the tubular reactor, quenched, and analyzed by gel permeation chromatography (GPC) to determine the polymer molecular weight and dispersity (*D*). Figure 3 details the results for the hourglass profile, in which the observed polymer molecular weights were in very close agreement with the targeted molecular weights. Additionally, narrow molecular weight distributions ($D < 1.10$) were achieved, establishing the success and precision of the polymerization. Flow ROP with a football and a bowtie architecture profile also resulted in good agreement of the experimental brush length with theory (Figures S13 and S19). It is worth noting that the bowtie profile contains a brush length step change of 75 repeat units, which occurred over 1.4% of macromonomer solution, highlighting the resolution that can be achieved without the loss of molecular weight control. Overall, method 1 has the advantage that it remains valid with the changes in the experimental setup (reagent concentration or reactor size), but it requires the rate law of the polymerization to be known.

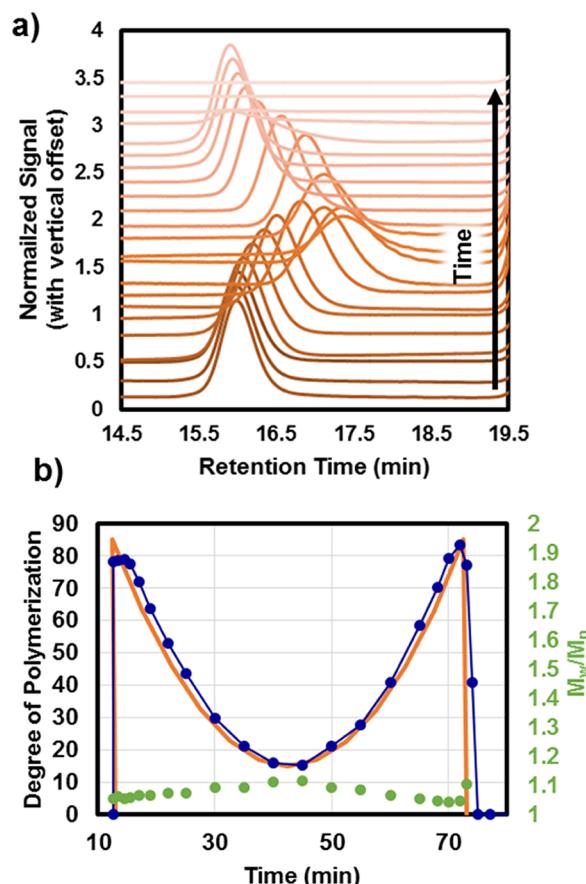


Figure 3. Demonstration of the ability of method 1 to relate a targeted architecture profile to the catalyst flow rate for the ROP of LA with an hourglass profile. (a) Stacked overlay of GPC chromatogram of aliquots exiting a single flow experiment showing the change in macromonomer molecular weight exiting the tubular reactor at different time points. (b) Data of the degree of polymerization and dispersity of the aliquots collected over time, revealing good agreement with the theoretical molecular weight predictions while maintaining a narrow molecular weight distribution. The orange line is the theoretical brush length profile, the blue dots are brush length calculated from GPC, and the green dots are molecular weight dispersity.

An alternative strategy, method 2, was developed when the polymerization rate law is not known. Method 2 is based on an empirical relationship between the degree of polymerization and catalyst loading for a fixed monomer concentration and a set range of catalyst concentrations. This relationship is experimentally determined in a single flow experiment where multiple different catalyst flow rates are run consecutively, and the resulting polymers are analyzed by GPC. By operating in the Poiseuille flow regime, where the Taylor dispersion is significant,^{93–101} step changes in the catalyst flow rate result in step changes in the degree of polymerization while maintaining narrow dispersities. Each flow rate becomes a data point for the relationship between the catalyst flow rate and the macromonomer length, and a function is fitted between all the data points (Figure 4a, Supporting Information Section 8). This function can then be used to convert any architecture profile (design parameter) to the ROP catalyst flow rate (process parameter). The ROP of VL catalyzed by 1,5,7-triazabicyclo[4.4.0]dec-5-ene (TBD) was performed to illustrate method 2. We obtained a linear relationship between the

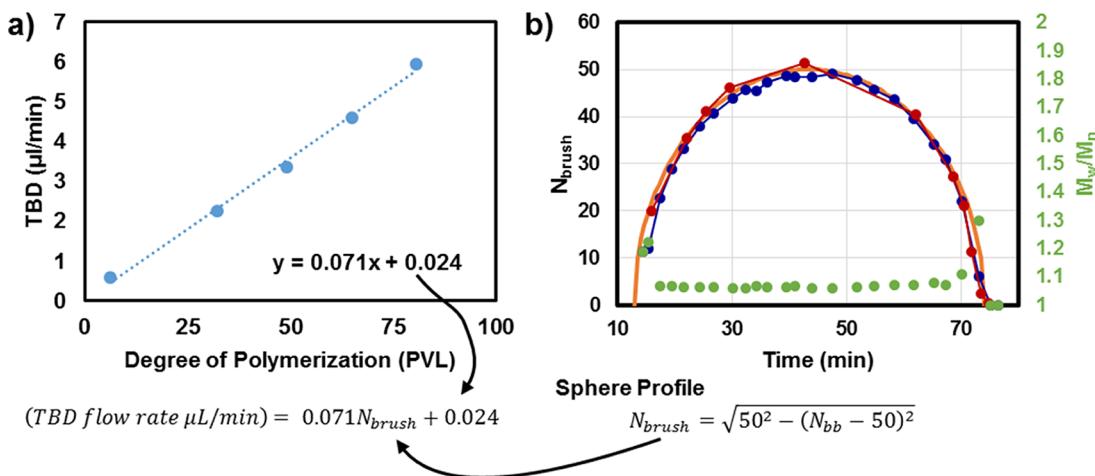


Figure 4. Demonstration of the ability of method 2 to relate a targeted architecture profile to the catalyst flow rate for the ROP of VL with a sphere profile. (a) Data for determining the empirical relationship between the degree of polymerization and catalyst loading via method 2. (b) Data of aliquots collected exiting the tubular reactor for the ROP of VL with a spherical profile. The orange line is the theoretical brush length profile, the red dots are brush length calculated from ^1H NMR, the blue dots are brush length calculated from GPC, and the green dots are molecular weight dispersity.

degree of polymerization and catalyst flow rates, but any function that describes the data can be used. The linear relationship was implemented to convert a spherical design profile into a TBD flow rate profile. The targeted and measured molecular weights were in very strong agreement, confirming the success of the methodology (Figure 4b). Overall, method 2 is faster and simpler to implement than method 1; however, any changes in the reactor, concentration, or monomer flow rates will require the determination of a new catalyst to degree of polymerization relationship.

Bottlebrush Synthesis and Characterization. Once the protocol for synthesizing macromonomers of various lengths with precise sequences was established, graft-through polymerization was added to the reactor setup to produce bottlebrush polymers. The synthesis of a polylactide (PLA) bottlebrush with an hourglass architecture was examined in detail by repeating the same experiment multiple times and rapidly quenching the ROMP reaction while simultaneously stopping the flow of macromonomers at different time points. The theoretical molecular weight of the bottlebrush polymer and the LA buildup (the unreacted LA that accumulates in the ROMP reaction vessel) can be calculated directly from the design equation and monomer molecular weights. ^1H nuclear magnetic resonance (NMR) of the aliquots provided the values for LA buildup and the residual levels of macromonomers present in solution, which were systematically below the detection limit (ROMP conversion $> 98\%$,⁶⁷ Supporting Information Section 6). GPC of the same aliquots showed that the bottlebrush polymers had narrow molecular weight distributions ($D < 1.1$), and most importantly, the measured molecular weights were in very close agreement with the theoretical molecular weights (Figure 5). The comparison of the experimental data with the corresponding theoretical values validates the success of the synthesis. The same hourglass bottlebrush was synthesized five times to yield 1.5 g of a bottlebrush polymer with a molecular weight standard deviation of 1.1%, showcasing the high reproducibility of the process ($N_{\text{BB}} = 200$, $M_{\text{n}} = 744 \text{ kg/mol}$, $D = 1.08$). The simplicity and versatility of the methodology were further illustrated by synthesizing bottlebrush polymers with football

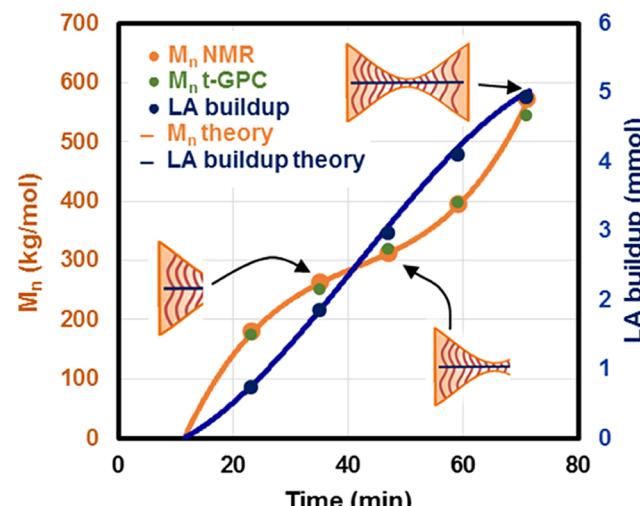


Figure 5. Molecular weight and LA buildup data for the synthesis of an hourglass bottlebrush. The data were obtained by performing the same synthesis multiple times and quenching the experiment at different time points for analysis. The lines are predictions calculated from the design equation and monomer molecular weights. The experimental data are represented by dots, in which M_{n} was obtained from ^1H NMR and triple detection GPC (t-GPC), and LA buildup was obtained from ^1H NMR.

and bowtie architecture profiles (Table 1). Finally, the chemical versatility was demonstrated by producing a poly(valerolactone) (PVL) bottlebrush with a spherical architecture ($N_{\text{BB}} = 100$, $M_{\text{n}} = 379 \text{ kg/mol}$, $D = 1.05$). In all cases, the bottlebrush polymers obtained had a narrow molecular weight distribution ($D < 1.15$), no residual macromonomers were detected (GPC and ^1H NMR), and the molecular weights and cyclic ester buildup matched the theoretical values.

Visualization of the bottlebrush polymers was done using AFM^{17,18,65,66,85,87,106–118} (Supporting Information Section 7). The images of hourglass bottlebrushes show two large ball ends connected by a thinner segment, which is consistent with the expectations of polymer conformations (Figure 6a). The measurement of the size of these molecular objects ($\sim 120 \text{ nm}$)

Table 1. Predicted and Experimental Data for the Synthesis of Architecture-Controlled Bottlebrush Polymers with $N_{BB} = 200$

entry	brush polymer	architecture	design and prediction		experimental data			
			LA/VL buildup (mmol)	M_n (kg/mol)	LA/VL buildup (mmol) ^a	conv. macro. ^a (%)	M_n (kg/mol) ^b	M_w/M_n ^b
1	PLA	hourglass	5.02	574	4.95	>98	546	1.07
2	PLA	football	4.60	699	4.55	>98	640	1.07
3	PLA	bowtie	4.60	698	4.58	>98	624	1.14
4	PVL	sphere	20.86	406	20.90	>98	379	1.05

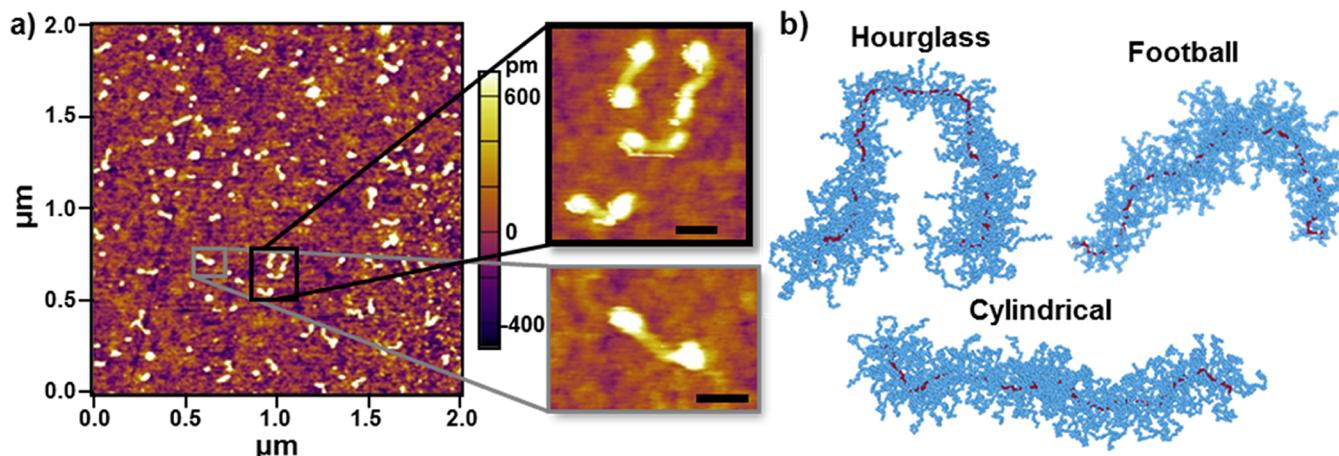
^aCalculated from conversion determined by ^1H NMR. ^bCalculated from t-GPC.

Figure 6. Images of architecture-controlled bottlebrush. (a) AFM images of hourglass bottlebrush polymers deposited on a silicon wafer with two zoomed-in images. The images depict polymers with two large ball ends connected by a thinner segment, consistent with the equilibrium conformations expected for an hourglass bottlebrush (scale bar = 50 nm). (b) Simulation snapshots of bottlebrush conformations for hourglass, football, and cylindrical architectures. Backbone beads are red and brush beads are blue. We are capable of visually observing the effect of different-length side chains on the bottlebrush conformations and note qualitative consistency between the hourglass conformation in simulation and the as-adsorbed conformations seen in AFM.

Table 2. Intrinsic Viscosity Data for Compositionally Identical Bottlebrushes ($N_{BB} = 200$)

architecture	experiment ^a				simulation	
	M_n ^b (kg/mol)	M_n/M_w ^b	$[\eta]$ (dL/g)	ratio ^c	$[\eta]$ dL/g	ratio ^c
hourglass	793	1.10	0.433	1.12	0.378	1.08
cylinder	768	1.03	0.387	1	0.349	1
football	811	1.05	0.318	0.82	0.282	0.81

^aTHF, 30 °C. ^bCalculated from t-GPC. ^cWith respect to cylinder $[\eta]$.

and comparison to the theoretical length show that some contraction of the core occurs, which is consistent with the backbone flexibility decreasing with increasing brush length.

Simulation is a powerful tool for understanding bottlebrush conformation,^{19,20,25–30,119–128} and our prior work has developed molecular simulation tools capable of quantitatively predicting the conformational structure of the synthesized bottlebrush molecules.⁸⁰ This same model is used to simulate architecture-controlled bottlebrush polymers (see Supporting Information Section 11). We show the snapshots of relaxed bottlebrush macromolecules calculated from Monte Carlo simulations in Figure 6b. The hourglass bottlebrush has short side chains in the center of the backbone contour, resulting in a higher flexibility of the backbone in the center. However, there is no apparent overall decrease in the backbone extension when compared to the cylindrical molecule. The snapshots of the hourglass architecture also qualitatively match with the AFM figures observed in Figure 6a. In contrast, the backbone at both ends of the football structure collapses toward the core because of the weak steric effects of the short side chains,

which results in a shorter molecular length when compared to the cylindrical bottlebrush.

We have previously demonstrated that the equilibrium conformations obtained by the simulation procedure are capable of reproducing quantitative trends observed in the viscometry of a wide range of cylindrical bottlebrushes.⁸⁰ This prior work illustrates that intrinsic viscosity can be used as a sensitive probe of architecture to establish the quantitative aspect of the simulation. Therefore, intrinsic viscosity measurements were performed on an hourglass, a cylindrical, and a football bottlebrush with identical chemical composition (same backbone degree of polymerization and same average brush degree of polymerization) and compared with intrinsic viscosity calculations from equilibrium simulations.^{129–132} Both methods showed that the hourglass bottlebrush has an intrinsic viscosity 10% larger when compared to a cylindrical bottlebrush. Conversely, the football bottlebrush has an intrinsic viscosity 20% smaller when compared to the cylindrical bottlebrush (Table 2). These results suggest that the football architecture takes a denser conformation in solution than the hourglass. This is further validated via the

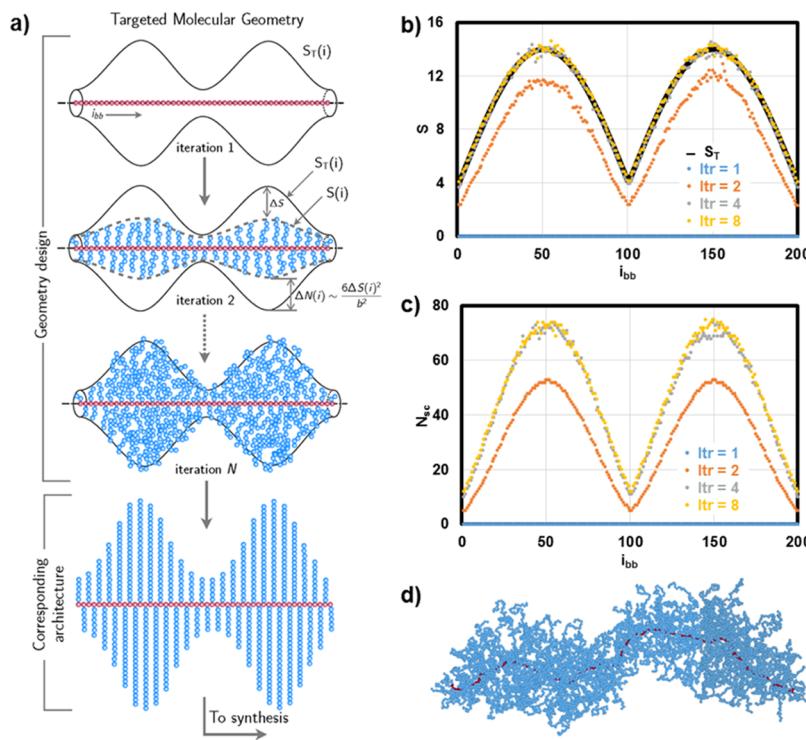


Figure 7. Inverse design scheme for bottlebrush geometry design. (a) Target width profile, the side-chain radius of gyration $S_T(i)$, is iteratively obtained by first taking a linear chain and adding $\Delta N(i)$ side-chain beads as a function of the index of the backbone monomer i . Subsequent iterations adjust this $\Delta N(i)$ based on the distance $\Delta S(i)$ of the width from the target, until convergence is obtained. The architectural information $N_{sc}(i)$ is an input to the synthesis, corresponding to the target molecular geometry. (b) Side-chain radius of gyration $S(i)$ as a function of iteration itr for an example geometry. Convergence to $S_T(i)$ is observed even after only four iterations. (c) Corresponding values of $N_{sc}(i)$ as a function of itr . We note that although there is a clear correspondence between the architecture $N_{sc}(i)$ and geometry $S_T(i)$, it is not proportional. (d) Simulation snapshot of the converged structure based on the inverse design protocol.

trends observed with their respective GPC elution times ($t_{\text{hourglass}} = 12.61 \pm 0.01$ min and $t_{\text{peak}}^{\text{football}} = 12.73 \pm 0.08$ min). Overall, the simulation data provide an additional independent confirmation of the success of the architecture-controlled synthesis.

Inverse Design of Molecular Geometry. The quantitative prediction of the simulation regarding the molecular conformation in solution provides a powerful method to connect the architecture profile and molecular geometry. This sets the basis of an inverse design protocol^{133–137} for calculating the bottlebrush architecture profile (process parameter) that must be synthesized to yield a specific molecular geometry (design parameter). We use the average radius of gyration $S = \langle R_{G,SC}^2 \rangle^{1/2}$ of a bottlebrush side chain as a descriptor of the bottlebrush “width”,⁸⁰ which can vary as a function of the position i along the bottlebrush backbone contour, $S(i)$. The desired axisymmetric design profile is defined by a target function $S_T(i)$. The inverse design process begins with a bare backbone chain, and we iteratively remove or add side-chain beads to approach this targeted profile. The number of beads added onto or removed from the side chain ΔN is determined based on an error function $E(i) = S_T(i) - S(i)$, via the equation $6E^2(i)/b^2 = \Delta N$, which when converged yields an architectural profile $N_{sc}(i)$ for the degree of polymerization for each side chain. Figure 7 shows this process, plotting both the contour-dependent $N_{sc}(i)$ and $R_{G,SC}(i)$, with a double-sphere profile as the targeted molecular geometry. The first iteration is a straight line at $N_{sc}(i)$, representing the initial linear homopolymer. Subsequent iterations add and subtract lengths from the bottlebrush side

chains, until a desired geometry $R_{G,SC}(i) \approx R_{G,T}(i)$ is found. The resulting architecture profile $N_{sc}(i)$ that leads to this geometric target $R_{G,T}(i)$ was subsequently used as an input to the automated flow synthesis, thereby allowing the engineering of macromolecules with controllable geometry (see [Supporting Information Section 6](#)). Significantly, we note that the desired molecular geometry and the architecture show similar trends but are not linearly proportional; this inverse design strategy enables the prediction of this nontrivial connection between the architecture and molecular geometry, which would otherwise be dependent not only on the side-chain length at a given location but also on the sequence of nearby side-chain lengths.

CONCLUSIONS

We have developed an automated methodology to synthesize macromolecules with 3D geometry. Our approach consists of performing a computer simulation to correlate molecular geometry to bottlebrush polymer architecture, followed by the synthesis of the predicted axisymmetric bottlebrush architecture. This synthetic achievement is enabled by the combination of the theoretical description provided by the simulation, the precise reactor engineering, and the living polymerizations employed. The simplicity, versatility, and productivity of using an automated flow reactor to control the polymer architecture will pave the way for rapid material design and lower the barrier for the synthesis of a large array of complex materials. Moreover, this methodology seeks to catalyze the establishment of new structure–function relationships in soft matter

with the aspiration of matching the specificity observed for biopolymers.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: [10.1021/acs.macromol.9b00845](https://doi.org/10.1021/acs.macromol.9b00845).

Scheme of the reactor setup correlated back to the real-world reactor setup; Mn versus time and rate versus lactide concentration for the ROP of lactide with various amounts of DBU and OH; DBU flow rate and GPC chromatograms for the football, bowtie, and hourglass ROP of lactide; GPC chromatograms and ¹H NMR spectra of the bottlebrush kinetic aliquots; cylinder, hourglass, and football profiles with identical composition; schematic of the inverse design to obtain brush profile function; height map and normalized height profiles of hourglass and football bottlebrush polymers; brush length results, GPC traces, and ¹H NMR spectra for the TBD flow rate sweep; TBD flow rate profile for a sphere bottlebrush; GPC chromatograms and ¹H NMR spectra for the sphere ROP of valerolactone; GPC chromatograms and ¹H NMR spectra for the synthesis of PVL bottlebrush; and data for the calculation of intrinsic viscosity ([PDF](#))

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Notes

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