



Cite this: DOI: 10.1039/c9nr08722b

A mini-review: emerging all-solid-state energy storage electrode materials for flexible devices

Yang Yang 

New technologies for future electronics such as personal healthcare devices and foldable smartphones require emerging developments in flexible energy storage devices as power sources. Besides the energy and power densities of energy devices, more attention should be paid to safety, reliability, and compatibility within highly integrated systems because they are almost in 24-hour real-time operation close to the human body. Thereupon, all-solid-state energy devices become the most promising candidates to meet these requirements. In this mini-review, the most recent research progress in all-solid-state flexible supercapacitors and batteries will be covered. The main focus of this mini-review is to summarize new materials development for all-solid-state flexible energy devices. The potential issues and perspectives regarding all-solid-state flexible energy device technologies will be highlighted.

Received 10th October 2019,
Accepted 31st December 2019

DOI: 10.1039/c9nr08722b

rsc.li/nanoscale

1. Introduction

An energy storage device is the critical technology in many defense and commercial applications where it is either directly used as a power source for transportation and electronics or integrated with other energy devices, such as thermoelectrics and piezoelectrics, to use energy in an efficient manner.^{1–3} In recent years, flexible energy storage has been taken into special consideration with increasing demands for wearable devices, including flexible displays, portable electronics, personal healthcare devices, and so forth.^{4–6} In terms of wearable electronics, the future will be miniaturized, integrated, and self-powered devices that are assembled in all-solid-state and can be utilized for a long period of time without safety and reliability issues.^{7–9} So far, some energy storage devices, for instance electrochemical capacitors (or supercapacitors), metal-ion batteries, and most recently rechargeable metal-air batteries, have been recognized to be the most practical and feasible technologies for all-solid-state energy storage.^{10–12} However, the main challenges in the current all-solid-state energy storage are still the low volumetric energy density, high internal resistance at the materials interfaces, poor mechanical durability, and controversial environmental concerns.^{13–16} To address these concerns, the following strategies have been suggested:

(i) The electrode materials should have a high areal and volumetric capacity, which represents their abilities to supply

high energy within a confined space.¹⁷ More specifically, emerging materials able to store energy based on mechanisms beyond double-layer storage and intercalation chemistry should be designed. In addition, some advanced features should be considered in electrode materials design such as additive-free and free-standing, which can significantly reduce the content of electrochemically inactive materials in the electrode.^{18–22}

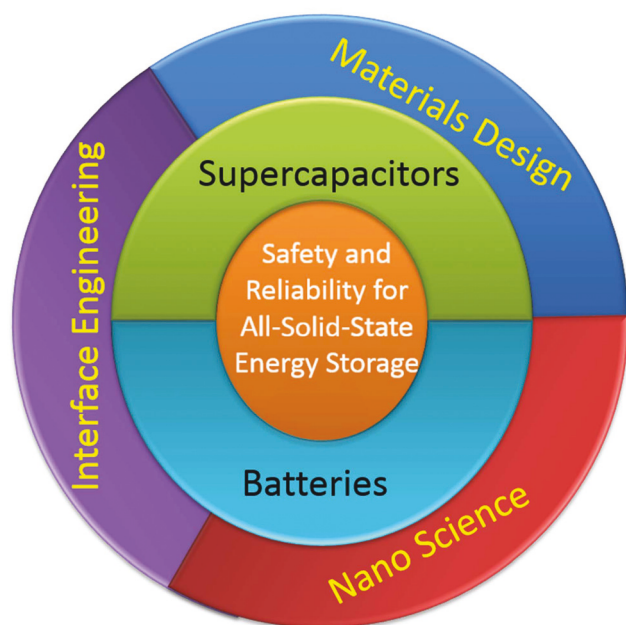
(ii) The electrode materials should have reduced resistance with the use of high conductivity materials or conducting components towards high-rate performance and less energy consumption by the internal resistance.²³ Experimentally, doping and hybridizing with other elements may improve the conductivity of electrode materials.²⁴ Computationally, data-driven simulation and machine-learning methods can be used to predict material compositions and structures to achieve the optimum electrode conductivity.²⁵

(iii) Advanced solid-state electrolyte films with high ionic conductivities should be designed.²⁶ The ionic conductivities of electrolyte films are always the critical point for solid-state energy storage. Similar to electrode materials, doping may solve the low ionic conductivity issue of the solid electrolyte.

(iv) Materials engineered at the nanoscale should be used to design new self-organized nanostructures that enable the release of the interfacial stress in all-solid-state devices.²⁷ Nano-engineering has been demonstrated to be very successful in boosting the material performance in renewable energy applications.^{28–30} Designing new electrode structures may solve the interfacial resistance and stability issues of solid-state devices.

(v) Environmentally benign and non-toxic materials should be utilized.³¹ Both electrode materials and solid electrolytes

NanoScience Technology Center, Department of Materials Science and Engineering,
Energy Conversion and Propulsion Cluster, University of Central Florida, 12424
Research Parkway Suite 423, Orlando, Florida 32826, USA.
E-mail: Yang.Yang@ucf.edu



Scheme 1 Illustration of the significance and methods to improve the safety and reliability of all-solid-state energy storage devices such as flexible supercapacitors and batteries by means of materials design, interface engineering, and nanoscience.

should be very safe and non-toxic in order to be used for flexible and personal devices. New electrode materials without metal dissolving and leaking should be considered for material design.

The detailed and specific reviews on electrode materials, solid-state electrolytes, and energy storage devices have been comprehensively discussed elsewhere.^{32–38} In this mini-review, therefore, recent advances in addressing the safety and reliability considerations of all-solid-state energy storage devices for wearable devices will be outlined (Scheme 1).

2. All-solid-state supercapacitors

Supercapacitors deliver a fast dynamic response, high power density, and long cyclability through non-faradaic and faradaic processes in electric double-layer capacitors (EDLCs) and pseudocapacitors, respectively.³⁹ The recent technological progress in the new nanostructured materials development, such as nano-carbons, other carbonaceous materials, and thin-film materials, enables all-solid-state supercapacitors to be reliable and environmentally benign for wearable devices.⁴⁰

2.1. Electrical double-layer capacitors

Non-faradaic EDLCs store energy in an ionic double-layer (Helmholtz layer) built upon the accumulation and electrostatic adsorption of electrolytic ions at the electrode/electrolyte interfaces, leading to extremely high power densities.⁴¹ The higher power density usually refers to a faster charging rate. On the other hand, the strong dependence of double-layered

charge storage on the surface areas of the electrodes limits the maximum possible number of the adsorbed ions at the EDLC interfaces, consequently resulting in low energy densities.⁴² This makes the EDLCs store insufficient energy for long-term utilization. A general strategy to improve the energy densities of EDLCs is to tremendously increase the electrode/electrolyte interfaces by using high surface conducting materials.⁴³ In this perspective, nano-carbon materials, including carbon nanotubes (CNTs), graphene, reduced graphene oxide (rGO), and nanoporous carbon fibers, and other carbon materials with hierarchically nanoporous structures have become the core competencies of all-solid-state EDLCs due to their good electrical conductivity, supreme mechanical strength, and highly engineerable surface area.^{44–48}

Creating nanopores in carbon-based EDLCs has been widely used as the most efficient approach to improve energy densities. Taking graphene as an example, in order to make the best use of its large theoretical surface area ($2630 \text{ m}^2 \text{ g}^{-1}$) and extraordinary electron mobility, different porous structures such as foams, aerogels, and sponges have been designed to overcome the aggregation problem.^{49–51} But these graphene architectures are of no help in creating additional active sites to increase the charge storage capacity. In addition to using porous architectures, directly creating nanopores in graphene has been suggested to be a fascinating approach to form hierarchically porous structures, leading to a dramatically improved capacitance.⁵² For instance, by integrating macropores, mesopores, and micropores into bamboo-like carbon fibers with a hierarchically porous structure, excellent reliability and outstanding capacitance have been achieved.⁵³ The usage of hierarchically porous carbons in EDLCs for wearable electronics is emerging but still controversial. This is because carbon materials usually have low densities, which require high material loading in limited space in order to achieve sufficiently high capacitance, therefore resulting in structural collapse.

Polyvinyl alcohol (PVA) gel has been widely used as a porous matrix to mix with ionically conducting agents such as phosphoric acid, KOH, other salts, and ionic liquids, forming solid polymer electrolytes for all-solid-state EDLCs. This is due to the attractive mechanical and electrical merits of PVA, including high stretchability, good strength, and controllable thickness and porosity.^{54–56} The most important feature of using PVA gel electrolytes is the reduced probability of electrolyte leakage, which makes wearable EDLCs tailorable (being cut) and safe at the device level.⁵⁷ Taking into account the fact that charge is stored at the surface or subsurface of EDLC electrodes, the following concerns are still unsolved in gel electrolyte based all-solid-state EDLCs: (i) since using high surface area nanoporous electrodes is necessary for high capacitance EDLCs, how to completely fill nanoscale pores with the gel and form a seamless interface. (ii) How to maintain sufficient strength to resist the intense mechanical impact in the gel electrolyte with weak molecular interactions. And these concerns may become more serious when a ceramic electrolyte is used because of the sluggish ion mobility across the poorly

contacted electrode/electrolyte interfaces.^{58–60} Most recently, Braun *et al.* reported a bottom-up method to completely infill gel electrolytes into porous electrodes (Fig. 1a).⁶¹ As a result, a 500 μm thick all-solid-state supercapacitor electrode composed of CNTs completely infilled with a PVA gel electrolyte was fabricated (Fig. 1b). In the rolling-up test, the CNT-based electrode did not show any microcracks, indicating exceptional mechanical properties (Fig. 1c). The excellent mechanical properties enable the CNT-based electrode to have 95% capacitance retention after 5000 bending cycles (Fig. 1d). The areal capacitance of the 500 μm thick supercapacitor electrode delivered an outstanding capacitance of 2662 mF cm^{-2} (Fig. 1e and f), which is superior to the current flexible supercapacitors (Fig. 1g).

Balancing energy and power densities in energy storage devices is always very challenging. Although recent advances have improved the energy densities of EDLCs, they are still known for providing a high power density for high energy consumption in a short period. Therefore, other high energy density energy devices such as pseudocapacitors and batteries have been developed in order to integrate with EDLCs and fill the functional gap (high energy density) in a solid-state device.

2.2. Pseudocapacitors

Beyond EDLCs, pseudocapacitors exhibit high energy densities derived from reversible faradaic redox reactions at the surface and subsurface of the materials, including some conducting

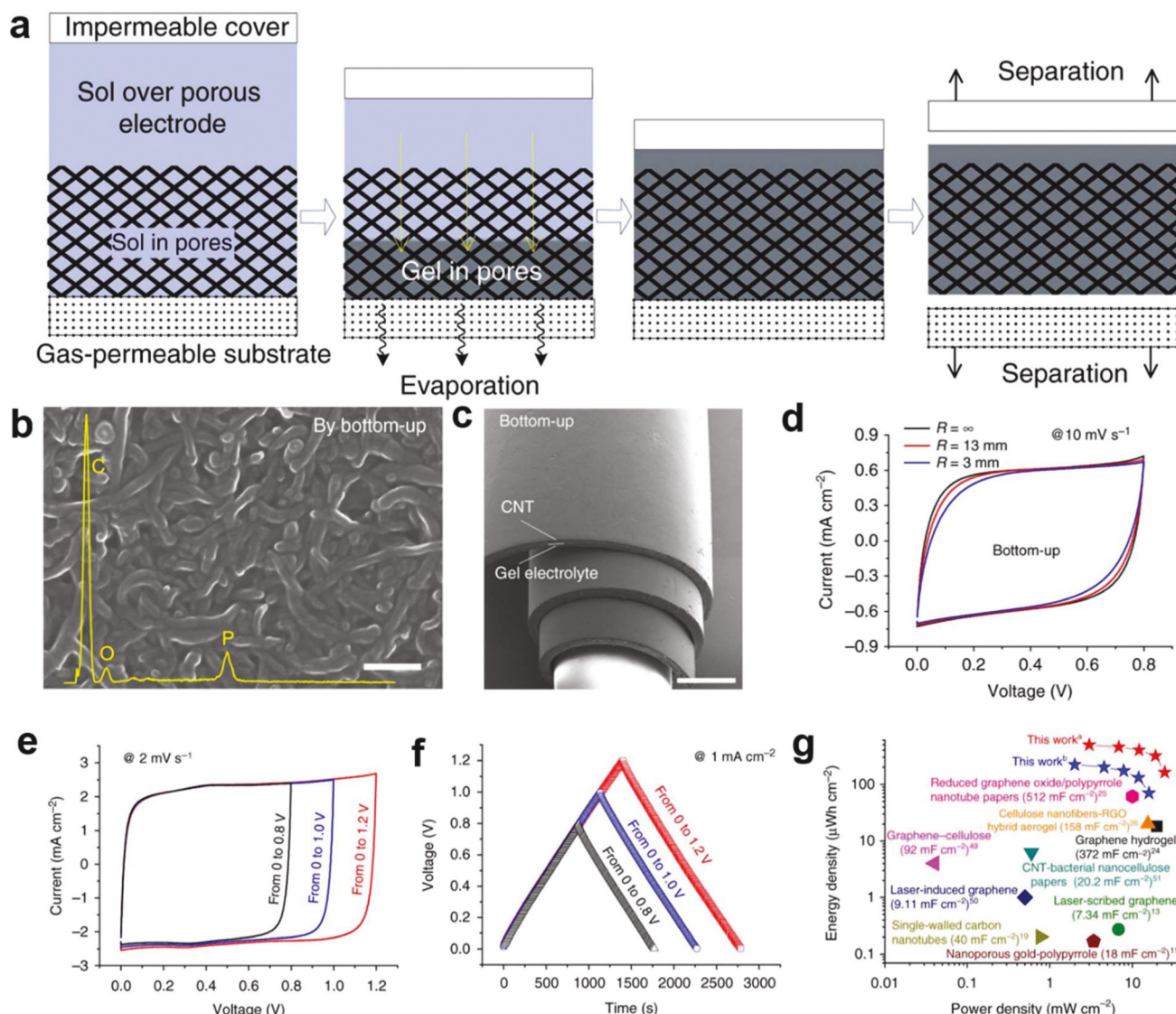


Fig. 1 (a) Schematic illustration of the bottom-up infilling method. Left to right: Sol casting on the porous electrode; the gel formation; a freestanding gel-filled electrode after substrate removal. (b) SEM image of the CNT-based electrodes infilled with the gel electrolyte. The scale bar is 200 nm. (c) The lack of microcracks in the 150 μm thick CNT-based electrodes infilled with the gel electrolyte tested by rolling-up over a glass tube with a radius of 0.5 mm. (d) Electrochemical performance of the CNT-based electrode supercapacitor. (e) CV and (f) GCD curves over different potential windows for the 500 μm thick CNT-based electrodes. (g) Areal Ragone plot. This figure has been adapted/reproduced from ref. 61 with permission from Springer Nature.

polymers (CPs) and transition metal compounds (TMCs).^{62–64} Although these CPs such as polyaniline (PANI), poly(3,4-ethylene dioxothiophene) (PEDOT), and polypyrrole (PPy) show great potential as active materials for pseudocapacitors, they are mostly used as conducting components in TMC-based electrodes rather than being used alone.^{65–67} This is because of their lower capacities than TMCs and low-density ($1\text{--}2\text{ g ml}^{-1}$) issue similar to carbon materials.^{68,69} The poor cycling stability of CPs caused by structural alteration and dissolving in organic electrolytes is another barrier to practical application.⁷⁰

Because of the insufficient flexibility and low conductivity, TMCs are always mixed with conducting materials such as carbons and CPs.⁷¹ In these TMC-based hybrid electrodes, carbons or CPs serve as conducting frameworks, which only contribute in a minor way to the overall capacitance as compared to TMCs.⁷² Thereupon, it is always a trade-off between the capacity and flexibility in pseudocapacitors using hybrid electrodes.^{73,74} It is necessary but very challenging to develop new techniques that can increase the loading amount of TMCs and meanwhile can maintain the mechanical strength without sacrificing the flexibility of hybrid electrodes. More recently, some advanced techniques, including growing or loading active materials on 3D metallic current collectors and forming highly porous nanostructured films, have been developed to increase the TMC loading in flexible and robust electrodes without using any carbon and CP additives.^{75–78} The self-adjusting capability of these 3D current collectors and porous films enables highly flexible and mechanically robust electrodes for wearable energy storage devices. Our group recently developed a freestanding NiFe oxyfluoride (NiFeOF) holey film (HF) using an electrochemical approach – electrodeposition followed by anodization (Fig. 2a–c).⁷⁹ The metallic NiFe alloy

filaments remained in the NiFeOF HF protected the nanoporous structure of the materials during bending and twisting tests (Fig. 2d and e). As a result, the as-prepared NiFeOF HF is freestanding and self-supporting, which can be used as an electrode without reliance on additives. The interconnected holey structure and hierarchical pores provide a high surface area for energy storage (Fig. 2f and g). Endowed with the good electrical conductivity and highly porous structure, the NiFeOF holey film delivered a maximum specific capacitance of 670 F cm^{-3} (Fig. 2h and i). Additionally, the NiFeOF HF presented exceptional performance stability under bending and long-term cycling conditions (Fig. 2j and k), showing supreme reliability for wearable electronic devices.

Although pseudocapacitors exhibit better energy densities than EDLCs and sometimes may show both acceptable energy densities and power densities, they may not really compete with EDLCs and batteries in terms of high power and energy, respectively.

3. Flexible batteries

Rechargeable batteries have been well investigated because of their capabilities to provide high energy density to power electronic devices for long-term utilization.⁸⁰ Metal-ion batteries, especially Li-ion batteries (LIBs), have been widely investigated for flexible batteries due to their abilities to be built in an all-solid-state construction.⁸¹ Although the energy storage mechanisms between LIBs and supercapacitors are significantly different, the basic concepts to develop flexible batteries are similar.⁸² To enhance the flexibility and conductivity of the battery electrodes, conducting additives such as carbon and carbonaceous materials are used to form composites with

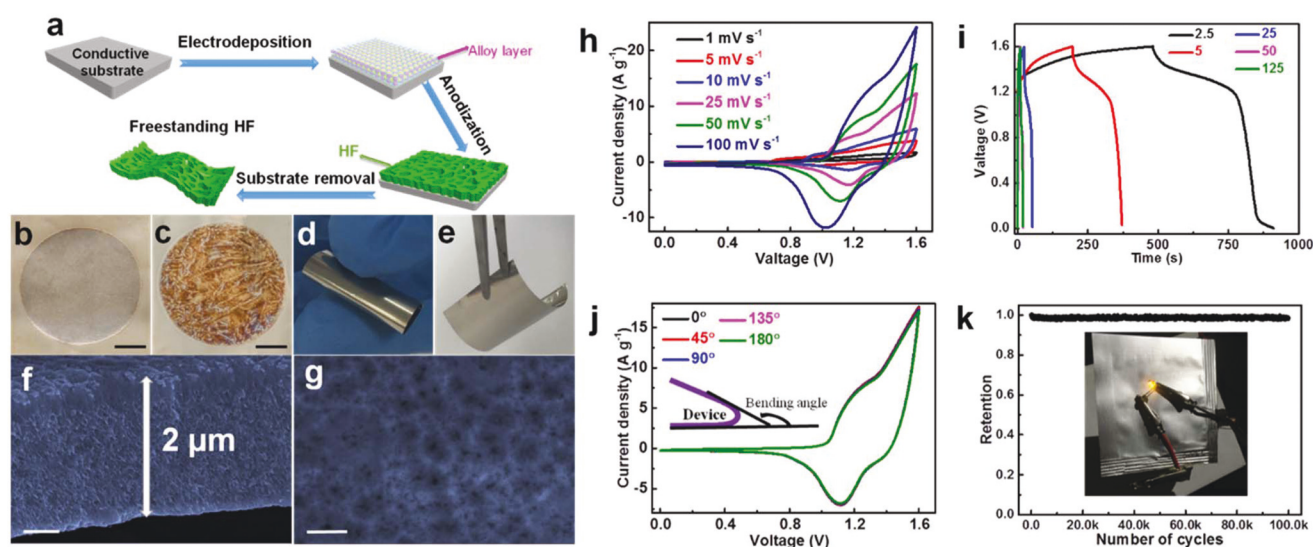


Fig. 2 (a) Schematic illustration of the electrochemical fabrication of the NiFeOF holey layer (HF). (b, c) Photos of the electrodeposited NiFe alloy and the anodized NiFeOF HF, respectively. The scale bars are 0.25 inch. (d, e) Excellent flexibility and mechanical robust feature of the freestanding HF. (f, g) Cross-sectional and top-view SEM images of HF. The scale bars in (f) and (g) denote 500 and 200 nm, respectively. (h) CV curves. (i) Charge/discharge profiles. (j) CV curves with different binding angles. (k) Cycling performance. This figure has been adapted/reproduced from ref. 79 with permission from John Wiley and Sons.

TMC-based active materials.⁸³ However, taking into account the nonaqueous electrolytes used in most rechargeable batteries, potential environmental and safety issues should be of concern to develop a human-compatible wearable device.⁸⁴

3.1. Li-ion batteries

The commercial and most typical LIBs are produced using layered materials as electrodes in which Li-ion intercalation/extraction processes are involved during discharge/charge cycling.⁸⁵ Li-Intercalation enables small volume expansion of active materials which ensures stable performance during long-term utilization, nevertheless, a low energy storage capacity is achieved.⁸⁶ Thereafter, novel electrode materials such as transition-metal based multi-valence materials based on other energy storage mechanisms such as alloying and conversion reactions have been developed to deliver much higher capacities than Li-intercalation electrodes, but the huge volume change becomes the major barrier to flexible batteries.⁸⁷ To address this issue, mechanically robust additives and self-adjusting nanostructures have been developed.^{88–90} Similar to flexible SCs, carbon materials and CPs are commonly used to form a flexible and high conductivity network to load electrode materials for flexible LIBs.⁷⁹ Mesoporous NiCo₂O₄ nanowire arrays coated with carbon textiles have been demonstrated to facilitate electron transport by directly connecting active materials to the current collectors and provide facile ion diffusion path by forming a mesoporous structure. Benefiting from these structural merits, binder-free NiCo₂O₄/carbon textile anodes exhibit high performance and excellent reliability.⁹¹ A polypyrrole@porous silicon hollow sphere (PPy@PHSi) composite exhibited excellent structural stability and electrochemical performance (high rate capability and outstanding cyclability) for flexible LIBs, owing to the synergism between the porous structure and the PPy coating. The porous structure of the shell buffered the volume change and reduced the internal stress, therefore facilitating Li⁺ diffusion in the porous electrodes. In addition, the PPy coating significantly enhanced the electrode conductivity and stabilized the structure.⁹² Mg-modified LiMnPO₄ nanofibers were mixed with a conducting carbon matrix in order to enhance the electrode conductivity, structural integrity, and flexibility. Therefore, a high capacity of 107 mA h g^{−1} was achieved at 5C, representing a superior rate performance for LiMnPO₄-based cathodes.⁹³ Flexible LIB full cells composed of Mn₂O₃ and LiMn₂O₄ nanowires as anodes and cathodes, respectively, have been developed without using any carbon supporting materials. The structure of one-dimensional nanowires provides a short Li⁺ transport path and volume flexibility during lithiation, leading to high rate performance.⁹⁴ Even though various advanced materials and new techniques have been conceived to fabricate flexible LIBs, it is still very challenging to achieve a balance between device flexibility (mechanical properties) and LIB performance (electrochemical properties such as power and energy densities) because of the electrochemically inactive components such as conducting additives and binders commonly used in the current flexible LIBs.

Safety issues related to organic electrolyte leakage, unsatisfactory packaging to withstand harsh utilization environments, and battery failure originating from either thermal runaway or metal dendrite growth are the chief concerns in view of practical applications of flexible LIBs in wearable and human-compatible electronic devices.⁹⁵ To address these battery safety risks, ceramic and novel polymer separators associated with thermal shutdown functions are of special importance in improving the safety and reliability of flexible LIBs.⁹⁶ A thermostable ceramic SiO₂-grafted PE separator prepared by electron beam irradiation showed a shrinkage ratio of only 20% even at 180 °C.⁹⁷ This separator also displayed an improved ionic conductivity because of the good wettability and electrochemical stability. Conventionally, most of the polymer separators are unstable at 120 °C and above when used in LIBs. To enhance thermal stability, a pure aluminum oxide nanowire-based membrane without any organic additives was developed as a bendable ceramic separator. At room temperature and 120 °C, LIBs fabricated with a ceramic separator showed a higher rate-performance and longer cyclability as compared to the conventional polymer separator.⁹⁸ Recently, ceramic nanowire fillers have been demonstrated to form ionically conducting networks in the polymer-based solid electrolyte. Dendrite growth threatens the safety of LIBs by piercing the separator and making the cell short. To address this issue, an aramid-PEO nanofiber composite has recently been demonstrated to show suppressed dendrite growth, a high modulus, and a high ionic conductivity. The small pores in the membranes were proved to eliminate the possibility of dendrite growth.

One of the most promising approaches to solve the safety issues in flexible LIBs is to utilize ionic liquids (ILs) in solid electrolytes due to their unique merits such as low melting point, almost no vapor pressure (non-volatile), high-temperature stability (non-flammable), and a wide electrochemical stability window.⁹⁹ When used in flexible LIBs, IL-based solid electrolytes do not release any toxic compounds such as HF if the cells are damaged and exposed to air.¹⁰⁰ Shahbazian-Yassar *et al.* recently reported an all-3D-printed LIB using an IL-based solid electrolyte (Fig. 3a–f).¹⁰¹ By adding nanosized ceramic fillers to the solid electrolyte, a continuous, thin, and dense layer was obtained between the porous electrolyte and the electrode. This is helpful to effectively reduce the interfacial resistance of all-solid-state LIBs. As a result, an ionic conductivity of 0.78×10^{-3} S cm^{−1} was achieved using a 3D printed solid electrolyte, leading to superior capacity and rate performance to the traditionally cast flexible LIBs (Fig. 3g–l).

LIBs surpass supercapacitors (EDLCs and pseudocapacitors) in terms of energy density, however they have the most serious safety and reliability concerns because of the metal dissolving/leaking and electrode material decomposition during improper usage such as over-charging/over-discharging.

3.2. Other metal-ion batteries

Besides LIBs, other flexible metal-ion batteries such as Na-ion, Mg-ion, and Al-ion batteries have been developed in recent years. Besides experiencing similar challenges to flexible LIBs,

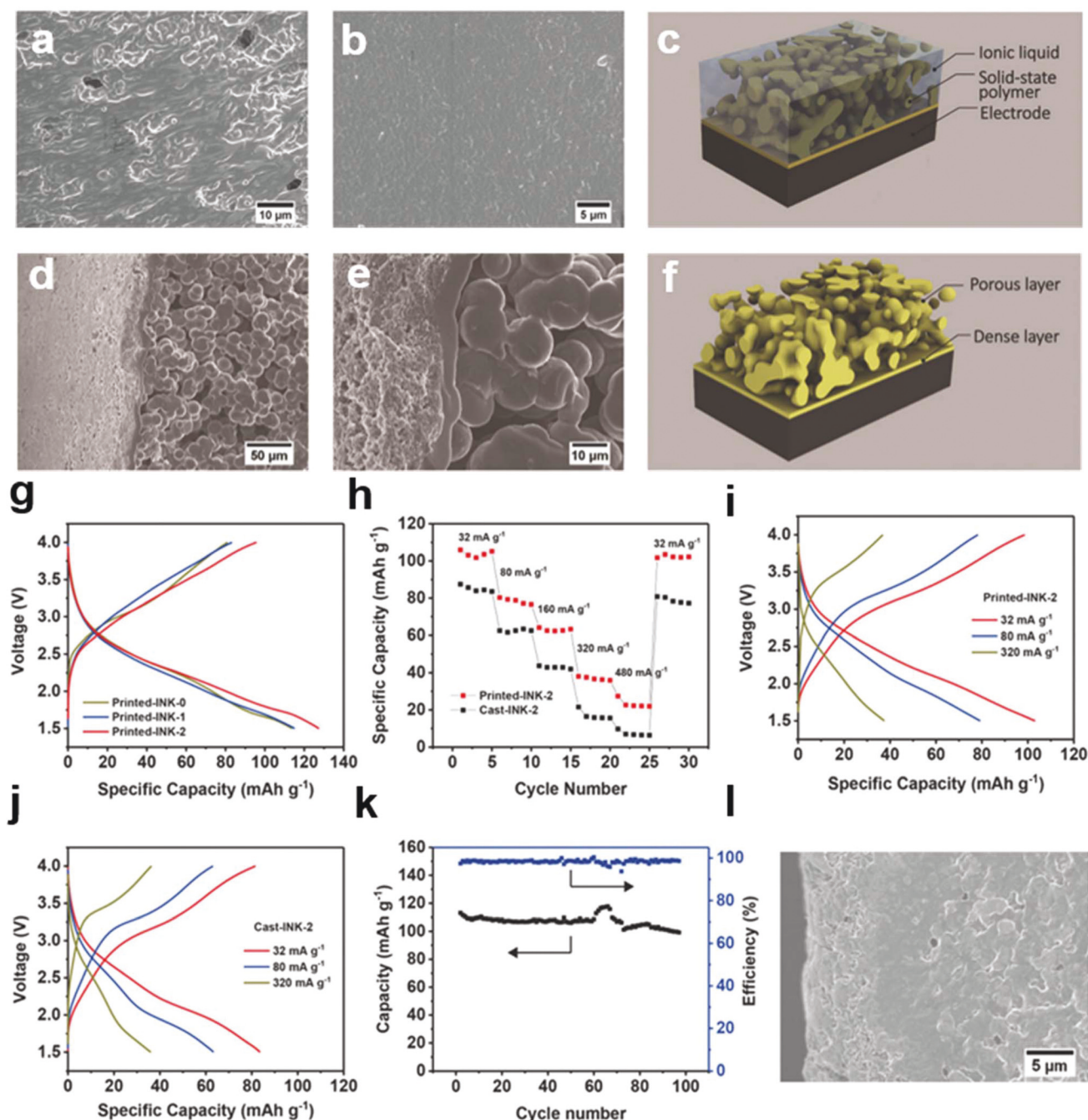


Fig. 3 (a, b) Cross-sectional and surface SEM images of the 3D printed electrolyte, respectively. (c) Schematic of the 3D printed solid electrolyte. (d) Cross-sectional SEM image of the 3D printed electrolyte after washing out the ionic liquid. (e) SEM image of the dense layer between the porous layer and the electrode. (f) Schematic of the bilayer structure of the 3D printed solid electrolyte. (g) First cycle charge–discharge profiles. (h) Rate profile. (i) Voltage profile of the 3D printed LIB cell. (j) Voltage profile of the cast LIB cell. (k) Cycling performance. (l) Cross-sectional SEM image of the 3D printed electrolyte after 100 cycles. This figure has been adapted/reproduced from ref. 101 with permission from John Wiley and Sons.

other flexible batteries face problems in terms of sluggish metal ion diffusion in the electrodes and limited options for electrode materials and electrolytes. A universal strategy to address these issues is to develop 3D electrodes with open channels in order to achieve the facilitated metal ion/cluster extraction and insertion from host materials to the electrolyte. However, the development of 3D electrodes for flexible Na-ion

batteries is still hampered by the intricate production methods and the relatively high cost of building blocks for 3D structures such as graphene and CNTs. Wu *et al.* designed a simple and low-cost electrospinning technique to wrap large-sized $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ with hierarchically 3D electronic channels for flexible Na-ion batteries.¹⁰² Such flexible electrodes exhibited outstanding electrolyte wettability, ultrafast electrical conductivity,

and high Na^+ ion diffusion coefficients, leading to a high reversible capacity of 116 mA h g^{-1} at 0.1C. Even at a high rate of 30C, a discharge capacity of 63 mA h g^{-1} was retained. Polymer-based solid electrolytes can boost the development of highly safe and flexible Mg-ion batteries because of great

merits, including greatly improved safety, high energy density, and structural flexibility. Recently, Cui *et al.* developed an *in situ* crosslinking reaction method to fabricate a novel polytetrahydrofuran-borate-based gel electrolyte coupled with glass fiber (Fig. 4a).¹⁰³ This gel electrolyte exhibited a reversible Mg

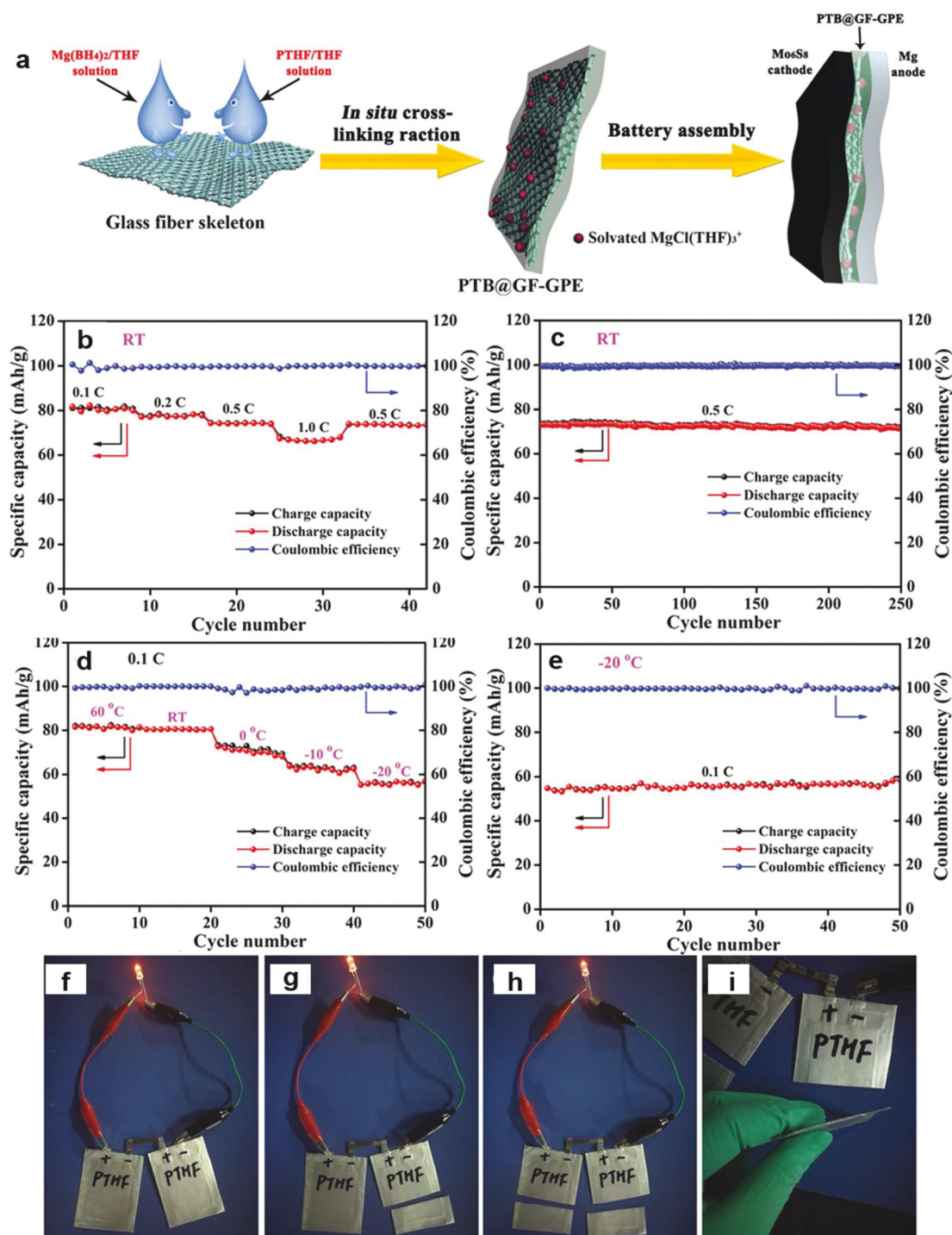


Fig. 4 (a) Schematic illustration of the preparation of $\text{MoS}_2/\text{PTB@GF-GPE}/\text{Mg}$ flexible Mg-ion battery cells. (b) C-Rate performance at room temperature (RT). (c) The RT cycling stability at 0.5C. (d) Temperature-dependent performance at 0.1C. (e) The cycling stability at 0.1C and at -20°C . (f–i) Flexible Mg-ion batteries in the cutting tests. This figure has been adapted/reproduced from ref. 103 with permission from John Wiley and Sons.

plating/stripping performance, high Mg^{2+} ionic conductivity, and remarkable Mg^{2+} ion transfer number (Fig. 4b and c). $\text{Mo}_6\text{S}_8/\text{Mg}$ batteries assembled with this gel electrolyte showed unprecedented electrochemical properties in a wide temperature range ($-20\text{ }^\circ\text{C}$ to $60\text{ }^\circ\text{C}$, Fig. 4d and e) and showed well-addressed safety issues without suffering from internal short-

circuit failure during the cutting test (Fig. 4f-i). Among all the metal-ion batteries beyond Li-ions, Al-ion batteries provide three-electron transfer during the Al^{3+}/Al redox reactions, as a result delivering a maximum gravimetric capacity of 2980 mA h g^{-1} . In addition, the price of Al is sufficiently low and stable in open air under ambient conditions for large-scale manufac-

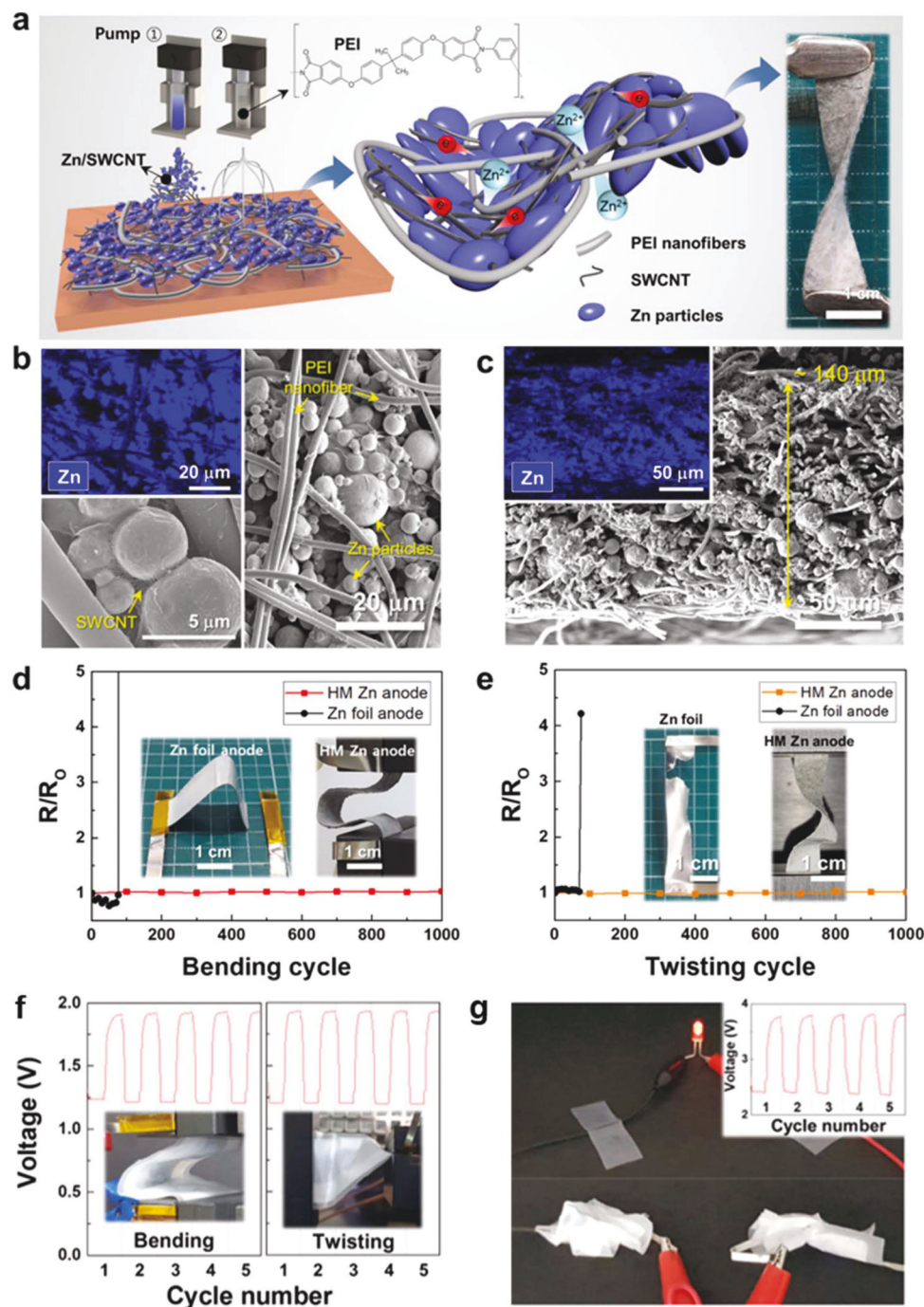


Fig. 5 (a) Schematic illustration of the electrospinning/electrospinning-assisted fabrication of a Zn anode. (b, c) Top-view and cross-sectional SEM images of a Zn anode, respectively. The insets show the EDS mapping of Zn. (d, e) The electrical resistance of the Zn anode as functions of bending and twisting, respectively. (f) Galvanostatic discharge/charge cycling during the bending and twisting tests. (g) The operation of an LED powered by flexible Zn-air batteries. The inset shows the galvanostatic discharge/charge cycling behavior of the flexible Zn-air batteries. This figure has been adapted/reproduced from ref. 112 with permission from American Chemical Society.

turing, which can significantly reduce the production cost of rechargeable batteries. However, the three-electron transfer process involved in the charge/discharge reactions, as well as the strong bonding between Al^{3+} and the host materials, makes Al-ion batteries slow in reaction kinetics. Most recently, our group designed a self-supported tin sulfide (SnS) porous film (PF), which was used as a flexible 3D cathode in Al-ion batteries, delivering a high specific capacity of 406 mA h g^{-1} .²⁴ A capacity decay rate of 0.03% per cycle is achieved, indicating good stability. This self-supported SnS film showed an outstanding electrochemical performance and stability during dynamic and static bending tests. The porous structure of SnS is beneficial for minimizing the volume expansion during charge/discharge, leading to improved structural stability when used as flexible 3D electrodes.

Although recent research progress has improved the kinetics for other metal-ion storage in electrode materials, more research efforts on enhancing the coulombic efficiency and further increasing the power/energy densities of the materials are still required.

3.3. Metal-air batteries

Metal-air batteries (MABs) store energy by the surface catalytic oxygen reduction and evolution reactions (ORR/OER) on the cathode side and redox reactions on the metal anode side.^{104–106} In principle, MABs deliver a much larger capacity to maintain a higher power density than that of LIBs. For the detailed background of the charging/discharging mechanism please refer to previous comprehensive review papers.^{37,107,108} In order to activate the catalytic reactions occurring on the cathodes, high surface area carbon-based bifunctional catalysts are preferred. Using a facile H_2 etching approach, Zhang *et al.* prepared a coaxial cable-like structure composed of a carbon fiber skeleton and nanostructured porous and defect-rich graphene skin.¹⁰⁹ By introducing more heteroatoms and defects as active sites, core-shell nanocarbons exhibited excellent OER/ORR activities and reliability in bending tests. Metal anodes play the central roles in determining the MAB energy density and cycle life because of metal (*e.g.* Al, Zn, Fe, and Mg) corrosion and the irreversible formation of discharge products on the electrode surfaces. More specifically, Zn anodes experience inhomogeneous deposition and the formation of dendrites when charging, leading to a quick loss of cyclability and potential safety issues caused by internal short circuits. Fe anodes suffer from surface passivation by iron hydroxide formed during the discharge process. Hence, increasing interest has been devoted to developing stable anode materials that are resistant to surface passivation and corrosion.¹¹⁰ Porous metals or alloys with 3D structures have been mostly used to suppress metal passivation and dendrite formation by eliminating the electrical field inhomogeneity during charging. However, relevant research progress in 3D metal anodes for flexible MABs is still very limited. Taking Zn-air batteries as an example, Zn anodes are most commonly fabricated by coating Zn particle slurries on the current collectors or directly using Zn foils.¹¹¹ Besides surface passivation by the discharge

product and internal short-circuit failure, these Zn anodes possess insufficient mechanical properties, leading to fatigue failure, mechanical rupture, and eventually losing the electrical contact under external deformation. Lee *et al.* developed a new class of flexible Zn-air batteries using a multifunctional heteronanomat (HM) architecture to address these issues. HM framework-supported electrodes were fabricated by a one-pot concurrent electro-spraying and electro-spinning process (Fig. 5a), forming 3D bicontinuous ion/electron transport channels in the electrodes (Fig. 5b and c). Benefiting from this unique structure, the HM-structured electrodes showed excellent electrical properties in deformation tests (Fig. 5d and e). In addition, when assembled into a flexible Zn-air battery cell, considerably improved mechanical properties and electrochemical rechargeability were achieved (Fig. 5f and g).¹¹²

MABs are the most promising energy devices that can be used to complement the traditional energy storage devices such as supercapacitors and metal-ion batteries because of their high energy densities and power densities. From the perspective of practical application, reducing the materials cost and designing the cell configuration should be focused on in future research in order to avoid the destruction of the electrode materials and electrolyte leakage.

4. Conclusions and outlook

Flexible electronic devices powered by bendable, foldable, and even cuttable energy devices are of significance for both defense and commercial utilization in the future. Safety should be placed in a priority position when developing flexible energy devices. Flexible supercapacitors are usually safe and provide high power but low energy, which cannot be used as power sources alone for electronic devices. Therefore, developing safe flexible batteries has become increasingly important. Making the energy devices all-solid-state could be a promising solution to the safety issue for flexible batteries. "All-solid-state" does not necessarily mean "rigid" or "brittle". Using 3D porous electrodes or developing advanced thin-film technology have been demonstrated to be very effective strategies for improving the flexibility of the device and meanwhile achieving an improved performance. Despite the increasing interest in flexible energy storage in recent years, there are many challenges and opportunities to be explored:

1. *Integration with other energy solutions:* Future flexible electronics should be minimized, foldable, and all-integrated and consume low energy. In order to provide a continuous energy supply, integrating with other energy sources such as solar or thermal energy could be a solution to the low energy density of state-of-the-art electrochemical energy storage devices. To this end, thin-film solar cells and thermoelectric devices could be used to generate electricity from external energy sources (sunlight, environment temperature change, *etc.*), which can provide energy to continuously charge flexible batteries or supercapacitors during day time. The integration of multiple components into a single device is in high demand for flexible

electronics, which requires the involvement of multiple disciplines in device design such as electrical engineering, industrial engineering, and so forth.

2. *Developing all bio-compatible materials for energy storage devices*: A special priority should be given to the development of bio-compatible materials for flexible energy devices. Besides safety concerns, the bio-compatibility should be paid more attention because all flexible devices would be used in the pocket, on the skin, or implanted in the human body. Therefore, soft materials, including polymers, gels, and soft biological materials could be considered in terms of bio-compatibility. However, it is always a big challenge to make a balance between energy density and safety when using “all-organic” soft materials in energy devices.

3. *Developing new nanomanufacturing techniques for energy device fabrication*: Inspired by the device design in the semiconductor industry, designing 3D interdigitated microstructures could be an effective way to address the mechanical and interfacial issues of flexible energy devices. In order to build seamless interfaces between the anode/solid-state-electrolyte and solid-state-electrolyte/cathode, cleanroom-based techniques such as patterning and lithography should be used for 3D interdigitated energy devices. And more importantly, new manufacturing processes such as additive manufacturing (3D and 4D printing) at the nanoscale could be introduced for the fabrication of 3D energy devices.

4. *Developing new energy harvesting devices*: Instead of electrochemical energy storage, new and emerging energy harvesting devices should be considered when designing flexible energy sources. If flexible electronic devices are used on the skin, there would be an opportunity to harvest the chemical energy from sweat or mechanical energy from vascular pulsation. Microstructured fuel cells and piezoelectric devices would play roles in converting other energy sources from the human body to electricity, which could power flexible electronics if the energy consumption for the devices is not high.

5. *Integrating data-driven prediction and machine-learning methods into materials design*: Materials design always involves a huge amount of experimental testing in order to identify an ideal composition or structure for the target performance. However, traditional trial-and-error methods are not efficient for new materials design. The emerging data-driven prediction and machine-learning methods should be integrated into experimental tests in order to predict the optimum material compositions and structures for the boosted performance toward solid-state energy storage.

6. *Advanced in situ/operando characterization*: Advanced *in situ/operando* characterization such as *in situ* X-ray techniques (XRD, XAS, etc.) and *in situ* microscopy (TEM, SEM, optical microscopy, etc.) is very helpful to identify the material growth mechanisms in the manufacturing process and material failure mechanisms in solid-state energy devices. Developing new *in situ/operando* techniques or integrating *in situ/operando* characterization into the materials design should be very crucial for solid-state devices.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Science Foundation under Grant No. CMMI-1851674 and the startup grant from the University of Central Florida.

References

- 1 X. Pu, W. Hu and Z. L. Wang, Toward Wearable Self-Charging Power Systems: The Integration of Energy-Harvesting and Storage Devices, *Small*, 2018, **14**(1), 1702817.
- 2 A. Gurung and Q. Qiao, Solar Charging Batteries: Advances, Challenges, and Opportunities, *Joule*, 2018, **2**(7), 1217–1230.
- 3 C. Wang, Y. Niu, J. Jiang, Y. Chen, H. Tian, R. Zhang, T. Zhou, J. Xia, Y. Pan and S. Wang, Hybrid thermoelectric battery electrode FeS₂ study, *Nano Energy*, 2018, **45**, 432–438.
- 4 D. Chen, Z. Lou, K. Jiang and G. Shen, Device Configurations and Future Prospects of Flexible/Stretchable Lithium-Ion Batteries, *Adv. Funct. Mater.*, 2018, **28**(51), 1805596.
- 5 G. Qian, X. Liao, Y. Zhu, F. Pan, X. Chen and Y. Yang, Designing Flexible Lithium-Ion Batteries by Structural Engineering, *ACS Energy Lett.*, 2019, **4**(3), 690–701.
- 6 L. Zeng, L. Qiu and H.-M. Cheng, Towards the practical use of flexible lithium ion batteries, *Energy Storage Mater.*, 2019, **23**, 434–438.
- 7 X. Cheng, J. Pan, Y. Zhao, M. Liao and H. Peng, Gel Polymer Electrolytes for Electrochemical Energy Storage, *Adv. Energy Mater.*, 2018, **8**(7), 1702184.
- 8 L. Qin, Q. Tao, A. El Ghazaly, J. Fernandez-Rodriguez, P. O. Å. Persson, J. Rosen and F. Zhang, High-Performance Ultrathin Flexible Solid-State Supercapacitors Based on Solution Processable Mo_{1.33}C MXene and PEDOT:PSS, *Adv. Funct. Mater.*, 2018, **28**(2), 1703808.
- 9 F. Zheng, M. Kotobuki, S. Song, M. O. Lai and L. Lu, Review on solid electrolytes for all-solid-state lithium-ion batteries, *J. Power Sources*, 2018, **389**, 198–213.
- 10 T. Nakamura, K. Amezawa, J. Kulisch, W. G. Zeier and J. Janek, Guidelines for All-Solid-State Battery Design and Electrode Buffer Layers Based on Chemical Potential Profile Calculation, *ACS Appl. Mater. Interfaces*, 2019, **11**(22), 19968–19976.
- 11 Z. Pan, J. Zhong, Q. Zhang, J. Yang, Y. Qiu, X. Ding, K. Nie, H. Yuan, K. Feng, X. Wang, G. Xu, W. Li, Y. Yao, Q. Li, M. Liu and Y. Zhang, Ultrafast All-Solid-State Coaxial Asymmetric Fiber Supercapacitors with a High

- Volumetric Energy Density, *Adv. Energy Mater.*, 2018, **8**(14), 1702946.
- 12 Q. Liu, Z. Chang, Z. Li and X. Zhang, Flexible Metal-Air Batteries: Progress, Challenges, and Perspectives, *Small Methods*, 2018, **2**(2), 1700231.
 - 13 Z. Pan, J. Yang, Q. Zhang, M. Liu, Y. Hu, Z. Kou, N. Liu, X. Yang, X. Ding, H. Chen, J. Li, K. Zhang, Y. Qiu, Q. Li, J. Wang and Y. Zhang, All-Solid-State Fiber Supercapacitors with Ultrahigh Volumetric Energy Density and Outstanding Flexibility, *Adv. Energy Mater.*, 2019, **9**(9), 1802753.
 - 14 J. Yue, M. Yan, Y.-X. Yin and Y.-G. Guo, Progress of the Interface Design in All-Solid-State Li-S Batteries, *Adv. Funct. Mater.*, 2018, **28**(38), 1707533.
 - 15 D. G. Seo and H. C. Moon, Mechanically Robust, Highly Ionic Conductive Gels Based on Random Copolymers for Bending Durable Electrochemical Devices, *Adv. Funct. Mater.*, 2018, **28**(14), 1706948.
 - 16 C. Zhao, L. Liu, X. Qi, Y. Lu, F. Wu, J. Zhao, Y. Yu, Y.-S. Hu and L. Chen, Solid-State Sodium Batteries, *Adv. Energy Mater.*, 2018, **8**(17), 1703012.
 - 17 J. Ma, J. Sung, J. Hong, S. Chae, N. Kim, S.-H. Choi, G. Nam, Y. Son, S. Y. Kim, M. Ko and J. Cho, Towards maximized volumetric capacity via pore-coordinated design for large-volume-change lithium-ion battery anodes, *Nat. Commun.*, 2019, **10**(1), 475.
 - 18 P. Sun, X. Zhao, R. Chen, T. Chen, L. Ma, Q. Fan, H. Lu, Y. Hu, Z. Tie, Z. Jin, Q. Xu and J. Liu, $\text{Li}_3\text{V}_2(\text{PO}_4)_3$ encapsulated flexible free-standing nanofabric cathodes for fast charging and long life-cycle lithium-ion batteries, *Nanoscale*, 2016, **8**(14), 7408–7415.
 - 19 R. Chen, X. Xue, J. Lu, T. Chen, Y. Hu, L. Ma, G. Zhu and Z. Jin, The dealloying–lithiation/delithiation–realloying mechanism of a breithauptite (NiSb) nanocrystal embedded nanofabric anode for flexible Li-ion batteries, *Nanoscale*, 2019, **11**(18), 8803–8811.
 - 20 R. Chen, X. Xue, Y. Hu, W. Kong, H. Lin, T. Chen and Z. Jin, Intermetallic SnSb nanodots embedded in carbon nanotubes reinforced nanofabric electrodes with high reversibility and rate capability for flexible Li-ion batteries, *Nanoscale*, 2019, **11**(28), 13282–13288.
 - 21 G. Zhu, L. Wang, H. Lin, L. Ma, P. Zhao, Y. Hu, T. Chen, R. Chen, Y. Wang, Z. Tie, J. Liu and Z. Jin, Walnut-Like Multicore–Shell MnO Encapsulated Nitrogen-Rich Carbon Nanocapsules as Anode Material for Long-Cycling and Soft-Packed Lithium-Ion Batteries, *Adv. Funct. Mater.*, 2018, **28**(18), 1800003.
 - 22 L. Ma, W. Zhang, L. Wang, Y. Hu, G. Zhu, Y. Wang, R. Chen, T. Chen, Z. Tie, J. Liu and Z. Jin, Strong Capillarity, Chemisorption, and Electrocatalytic Capability of Crisscrossed Nanostraws Enabled Flexible, High-Rate, and Long-Cycling Lithium–Sulfur Batteries, *ACS Nano*, 2018, **12**(5), 4868–4876.
 - 23 D. J. Noelle, M. Wang, A. V. Le, Y. Shi and Y. Qiao, Internal resistance and polarization dynamics of lithium-ion batteries upon internal shorting, *Appl. Energy*, 2018, **212**, 796–808.
 - 24 K. Liang, L. Ju, S. Koul, A. Kushima and Y. Yang, Self-Supported Tin Sulfide Porous Films for Flexible Aluminum-Ion Batteries, *Adv. Energy Mater.*, 2019, **9**(2), 1802543.
 - 25 Z. Li, W. Niu, Z. Yang, N. Zaman, W. Samarakoon, M. Wang, A. Kara, M. Lucero, M. V. Vyas, H. Cao, H. Zhou, G. E. Sterbinsky, Z. Feng, Y. Du and Y. Yang, Stabilizing atomic Pt with trapped interstitial F in alloyed PtCo nanosheets for high-performance zinc-air batteries, *Energy Environ. Sci.*, 2020, DOI: 10.1039/C9EE02657F.
 - 26 Z. Gao, H. Sun, L. Fu, F. Ye, Y. Zhang, W. Luo and Y. Huang, Promises, Challenges, and Recent Progress of Inorganic Solid-State Electrolytes for All-Solid-State Lithium Batteries, *Adv. Mater.*, 2018, **30**(17), 1705702.
 - 27 W. Li, Y. Xia, J. Zhu and H. Luo, State-of-Charge Dependence of Mechanical Response of Lithium-Ion Batteries: A Result of Internal Stress, *J. Electrochem. Soc.*, 2018, **165**(7), A1537–A1546.
 - 28 L. Guo, Z. Yang, K. Marcus, Z. Li, B. Luo, L. Zhou, X. Wang, Y. Du and Y. Yang, $\text{MoS}_2/\text{TiO}_2$ heterostructures as nonmetal plasmonic photocatalysts for highly efficient hydrogen evolution, *Energy Environ. Sci.*, 2018, **11**(1), 106–114.
 - 29 L. Shi, W. Zhou, Z. Li, S. Koul, A. Kushima and Y. Yang, Periodically Ordered Nanoporous Perovskite Photoelectrode for Efficient Photoelectrochemical Water Splitting, *ACS Nano*, 2018, **12**(6), 6335–6342.
 - 30 J. Chang, G. Wang, A. Belharsa, J. Ge, W. Xing and Y. Yang, Stable $\text{Fe}_2\text{P}_2\text{S}_6$ Nanocrystal Catalyst for High-Efficiency Water Electrolysis, *Small Methods*, 2019, DOI: 10.1002/smt.201900632.
 - 31 Y.-Y. Wang, W.-Y. Diao, C.-Y. Fan, X.-L. Wu and J.-P. Zhang, Benign Recycling of Spent Batteries towards All-Solid-State Lithium Batteries, *Chem. – Eur. J.*, 2019, **25**(38), 8975–8981.
 - 32 D. R. MacFarlane, M. Forsyth, P. C. Howlett, M. Kar, S. Passerini, J. M. Pringle, H. Ohno, M. Watanabe, F. Yan, W. Zheng, S. Zhang and J. Zhang, Ionic liquids and their solid-state analogues as materials for energy generation and storage, *Nat. Rev. Mater.*, 2016, **1**(2), 15005.
 - 33 J. F. M. Oudenhoven, L. Baggetto and P. H. L. Notten, All-Solid-State Lithium-Ion Microbatteries: A Review of Various Three-Dimensional Concepts, *Adv. Energy Mater.*, 2011, **1**(1), 10–33.
 - 34 P. G. Bruce, S. A. Freunberger, L. J. Hardwick and J.-M. Tarascon, Li-O_2 and Li-S batteries with high energy storage, *Nat. Mater.*, 2011, **11**, 19.
 - 35 P. Yang and W. Mai, Flexible solid-state electrochemical supercapacitors, *Nano Energy*, 2014, **8**, 274–290.
 - 36 A. Eftekhari, L. Li and Y. Yang, Polyaniline supercapacitors, *J. Power Sources*, 2017, **347**, 86–107.
 - 37 W. Niu and Y. Yang, Graphitic Carbon Nitride for Electrochemical Energy Conversion and Storage, *ACS Energy Lett.*, 2018, **3**(11), 2796–2815.

- 38 K. Liang, L. Li and Y. Yang, Inorganic Porous Films for Renewable Energy Storage, *ACS Energy Lett.*, 2017, **2**(2), 373–390.
- 39 J. R. Miller and P. Simon, Electrochemical Capacitors for Energy Management, *Science*, 2008, **321**(5889), 651–652.
- 40 P. Simon, Y. Gogotsi and B. Dunn, Where Do Batteries End and Supercapacitors Begin?, *Science*, 2014, **343**(6176), 1210–1211.
- 41 J. W. Long, D. Bélanger, T. Brousse, W. Sugimoto, M. B. Sassin and O. Crosnier, Asymmetric electrochemical capacitors—Stretching the limits of aqueous electrolytes, *MRS Bull.*, 2011, **36**(7), 513–522.
- 42 A. Burke, Ultracapacitors: why, how, and where is the technology, *J. Power Sources*, 2000, **91**(1), 37–50.
- 43 L. Zhang, X. Hu, Z. Wang, F. Sun and D. G. Dorrell, A review of supercapacitor modeling, estimation, and applications: A control/management perspective, *Renewable Sustainable Energy Rev.*, 2018, **81**, 1868–1878.
- 44 T. Lv, M. Liu, D. Zhu, L. Gan and T. Chen, Nanocarbon-Based Materials for Flexible All-Solid-State Supercapacitors, *Adv. Mater.*, 2018, **30**(17), 1705489.
- 45 L. Chen, D. Li, L. Chen, P. Si, J. Feng, L. Zhang, Y. Li, J. Lou and L. Ci, Core-shell structured carbon nanofibers yarn@polypyrrole@graphene for high performance all-solid-state fiber supercapacitors, *Carbon*, 2018, **138**, 264–270.
- 46 L. Feng, K. Wang, X. Zhang, X. Sun, C. Li, X. Ge and Y. Ma, Flexible Solid-State Supercapacitors with Enhanced Performance from Hierarchically Graphene Nanocomposite Electrodes and Ionic Liquid Incorporated Gel Polymer Electrolyte, *Adv. Funct. Mater.*, 2018, **28**(4), 1704463.
- 47 R. Hu, J. Zhao, G. Zhu and J. Zheng, Fabrication of flexible free-standing reduced graphene oxide/polyaniline nanocomposite film for all-solid-state flexible supercapacitor, *Electrochim. Acta*, 2018, **261**, 151–159.
- 48 Y. Zhang, H. Jiang, Q. Wang, J. Zheng and C. Meng, Kelp-derived three-dimensional hierarchical porous N, O-doped carbon for flexible solid-state symmetrical supercapacitors with excellent performance, *Appl. Surf. Sci.*, 2018, **447**, 876–885.
- 49 Z. Wu, C. Xu, C. Ma, Z. Liu, H.-M. Cheng and W. Ren, Synergistic Effect of Aligned Graphene Nanosheets in Graphene Foam for High-Performance Thermally Conductive Composites, *Adv. Mater.*, 2019, **31**(19), 1900199.
- 50 J. Mao, J. Iocozzia, J. Huang, K. Meng, Y. Lai and Z. Lin, Graphene aerogels for efficient energy storage and conversion, *Energy Environ. Sci.*, 2018, **11**(4), 772–799.
- 51 N. Yousefi, X. Lu, M. Elimelech and N. Tufenkji, Environmental performance of graphene-based 3D macrostructures, *Nat. Nanotechnol.*, 2019, **14**(2), 107–119.
- 52 H. Wang, L. Zhi, K. Liu, L. Dang, Z. Liu, Z. Lei, C. Yu and J. Qiu, Thin-Sheet Carbon Nanomesh with an Excellent Electrocapacitive Performance, *Adv. Funct. Mater.*, 2015, **25**(34), 5420–5427.
- 53 Y. Sun, R. B. Sills, X. Hu, Z. W. Seh, X. Xiao, H. Xu, W. Luo, H. Jin, Y. Xin, T. Li, Z. Zhang, J. Zhou, W. Cai, Y. Huang and Y. Cui, A Bamboo-Inspired Nanostructure Design for Flexible, Foldable, and Twistable Energy Storage Devices, *Nano Lett.*, 2015, **15**(6), 3899–3906.
- 54 S. B. Aziz, T. J. Woo, M. F. Z. Kadir and H. M. Ahmed, A conceptual review on polymer electrolytes and ion transport models, *J. Sci.: Adv. Mater. Devices*, 2018, **3**(1), 1–17.
- 55 S. A. Alexandre, G. G. Silva, R. Santamaría, J. P. C. Trigueiro and R. L. Lavall, A highly adhesive PIL/IL gel polymer electrolyte for use in flexible solid state supercapacitors, *Electrochim. Acta*, 2019, **299**, 789–799.
- 56 R. Na, N. Lu, S. Zhang, G. Huo, Y. Yang, C. Zhang, Y. Mu, Y. Luo and G. Wang, Facile synthesis of a high-performance, fire-retardant organic gel polymer electrolyte for flexible solid-state supercapacitors, *Electrochim. Acta*, 2018, **290**, 262–272.
- 57 B. Xie, C. Yang, Z. Zhang, P. Zou, Z. Lin, G. Shi, Q. Yang, F. Kang and C.-P. Wong, Shape-Tailorable Graphene-Based Ultra-High-Rate Supercapacitor for Wearable Electronics, *ACS Nano*, 2015, **9**(6), 5636–5645.
- 58 S. J. Kwon, T. Kim, B. M. Jung, S. B. Lee and U. H. Choi, Multifunctional Epoxy-Based Solid Polymer Electrolytes for Solid-State Supercapacitors, *ACS Appl. Mater. Interfaces*, 2018, **10**(41), 35108–35117.
- 59 Y.-P. Liu, X.-H. Qi, L. Li, S.-H. Zhang and T. Bi, MOF-derived PPy/carbon-coated copper sulfide ceramic nanocomposite as high-performance electrode for supercapacitor, *Ceram. Int.*, 2019, **45**(14), 17216–17223.
- 60 B. Qin, Y. Han, Y. Ren, D. Sui, Y. Zhou, M. Zhang, Z. Sun, Y. Ma and Y. Chen, A Ceramic-Based Separator for High-Temperature Supercapacitors, *Energy Technol.*, 2018, **6**(2), 306–311.
- 61 X. Li, J. Shao, S.-K. Kim, C. Yao, J. Wang, Y.-R. Miao, Q. Zheng, P. Sun, R. Zhang and P. V. Braun, High energy flexible supercapacitors formed via bottom-up infilling of gel electrolytes into thick porous electrodes, *Nat. Commun.*, 2018, **9**(1), 2578.
- 62 C. Zhang, Super pseudocapacitors, *Nat. Energy*, 2018, **3**(12), 1019–1019.
- 63 Y. Wang, W. Li, Y. Guo, J. Cao, I. Murtaza, A. Shuja, Y. He and H. Meng, Recombination Strategy for Processable Ambipolar Electroactive Polymers in Pseudocapacitors, *Macromolecules*, 2018, **51**(18), 7350–7359.
- 64 G. Wang, S. Oswald, M. Löffler, K. Müllen and X. Feng, Beyond Activated Carbon: Graphite-Cathode-Derived Li-Ion Pseudocapacitors with High Energy and High Power Densities, *Adv. Mater.*, 2019, **31**(14), 1807712.
- 65 M. Boota and Y. Gogotsi, MXene—Conducting Polymer Asymmetric Pseudocapacitors, *Adv. Energy Mater.*, 2019, **9**(7), 1802917.
- 66 G. S. Gund, J. H. Park, R. Harpalsinh, M. Kota, J. H. Shin, T.-I. Kim, Y. Gogotsi and H. S. Park, MXene/Polymer Hybrid Materials for Flexible AC-Filtering Electrochemical Capacitors, *Joule*, 2019, **3**(1), 164–176.

- 67 R. Manikandan, C. J. Raj, M. Rajesh, B. C. Kim, J. Y. Sim and K. H. Yu, Electrochemical Behaviour of Lithium, Sodium and Potassium Ion Electrolytes in a $\text{Na}_{0.33}\text{V}_2\text{O}_5$ Symmetric Pseudocapacitor with High Performance and High Cyclic Stability, *ChemElectroChem*, 2018, 5(1), 101–111.
- 68 Z. Zhang, M. Liao, H. Lou, Y. Hu, X. Sun and H. Peng, Conjugated Polymers for Flexible Energy Harvesting and Storage, *Adv. Mater.*, 2018, 30(13), 1704261.
- 69 S. Cho, M. Kim and J. Jang, Screen-Printable and Flexible RuO_2 Nanoparticle-Decorated PEDOT:PSS/Graphene Nanocomposite with Enhanced Electrical and Electrochemical Performances for High-Capacity Supercapacitor, *ACS Appl. Mater. Interfaces*, 2015, 7(19), 10213–10227.
- 70 Z. Zhou and X.-F. Wu, High-performance porous electrodes for pseudosupercapacitors based on graphene-beaded carbon nanofibers surface-coated with nanostructured conducting polymers, *J. Power Sources*, 2014, 262, 44–49.
- 71 Z. Lv, Y. Luo, Y. Tang, J. Wei, Z. Zhu, X. Zhou, W. Li, Y. Zeng, W. Zhang, Y. Zhang, D. Qi, S. Pan, X. J. Loh and X. Chen, Editable Supercapacitors with Customizable Stretchability Based on Mechanically Strengthened Ultralong MnO_2 Nanowire Composite, *Adv. Mater.*, 2018, 30(2), 1704531.
- 72 Z. Ling, A. Harvey, D. McAteer, I. J. Godwin, B. Szydłowska, A. Griffin, V. Vega-Mayoral, Y. Song, A. Seral-Ascaso, V. Nicolosi and J. Coleman, Quantifying the Role of Nanotubes in Nano:Nano Composite Supercapacitor Electrodes, *Adv. Energy Mater.*, 2018, 8(8), 1702364.
- 73 W. Yang, Z. Gao, J. Ma, X. Zhang, J. Wang and J. Liu, Hierarchical NiCo_2O_4 @ NiO core-shell hetero-structured nanowire arrays on carbon cloth for a high-performance flexible all-solid-state electrochemical capacitor, *J. Mater. Chem. A*, 2014, 2(5), 1448–1457.
- 74 M. S. Javed, S. Dai, M. Wang, D. Guo, L. Chen, X. Wang, C. Hu and Y. Xi, High performance solid state flexible supercapacitor based on molybdenum sulfide hierarchical nanospheres, *J. Power Sources*, 2015, 285, 63–69.
- 75 G. Zhu, Z. He, J. Chen, J. Zhao, X. Feng, Y. Ma, Q. Fan, L. Wang and W. Huang, Highly conductive three-dimensional MnO_2 -carbon nanotube-graphene-Ni hybrid foam as a binder-free supercapacitor electrode, *Nanoscale*, 2014, 6(2), 1079–1085.
- 76 X. Peng, L. Peng, C. Wu and Y. Xie, Two dimensional nanomaterials for flexible supercapacitors, *Chem. Soc. Rev.*, 2014, 43(10), 3303–3323.
- 77 Y. Yang, H. Fei, G. Ruan and J. M. Tour, Porous Cobalt-Based Thin Film as a Bifunctional Catalyst for Hydrogen Generation and Oxygen Generation, *Adv. Mater.*, 2015, 27(20), 3175–3180.
- 78 Y. Yang, H. Fei, G. Ruan, L. Li, G. Wang, N. D. Kim and J. M. Tour, Carbon-Free Electrocatalyst for Oxygen Reduction and Oxygen Evolution Reactions, *ACS Appl. Mater. Interfaces*, 2015, 7(37), 20607–20611.
- 79 K. Liang, K. Marcus, Z. Yang, L. Zhou, H. Pan, Y. Bai, Y. Du, M. H. Engelhard and Y. Yang, Freestanding NiFe Oxyfluoride Holey Film with Ultrahigh Volumetric Capacitance for Flexible Asymmetric Supercapacitors, *Small*, 2018, 14(3), 1702295.
- 80 M. Winter, B. Barnett and K. Xu, Before Li Ion Batteries, *Chem. Rev.*, 2018, 118(23), 11433–11456.
- 81 J.-L. Shi, D.-D. Xiao, M. Ge, X. Yu, Y. Chu, X. Huang, X.-D. Zhang, Y.-X. Yin, X.-Q. Yang, Y.-G. Guo, L. Gu and L.-J. Wan, High-Capacity Cathode Material with High Voltage for Li-Ion Batteries, *Adv. Mater.*, 2018, 30(9), 1705575.
- 82 B. K. Peters, K. X. Rodriguez, S. H. Reisberg, S. B. Beil, D. P. Hickey, Y. Kawamata, M. Collins, J. Starr, L. Chen, S. Udyavara, K. Klunder, T. J. Gorey, S. L. Anderson, M. Neurock, S. D. Minter and P. S. Baran, Scalable and safe synthetic organic electroreduction inspired by Li-ion battery chemistry, *Science*, 2019, 363(6429), 838–845.
- 83 J. B. Goodenough, How we made the Li-ion rechargeable battery, *Nat. Electron.*, 2018, 1(3), 204–204.
- 84 M. R. Palacin, Understanding ageing in Li-ion batteries: a chemical issue, *Chem. Soc. Rev.*, 2018, 47(13), 4924–4933.
- 85 P. K. Nayak, E. M. Erickson, F. Schipper, T. R. Penki, N. Munichandraiah, P. Adelhelm, H. Sclar, F. Amalraj, B. Markovsky and D. Aurbach, Review on Challenges and Recent Advances in the Electrochemical Performance of High Capacity Li- and Mn-Rich Cathode Materials for Li-Ion Batteries, *Adv. Energy Mater.*, 2018, 8(8), 1702397.
- 86 C. Zhan, T. Wu, J. Lu and K. Amine, Dissolution, migration, and deposition of transition metal ions in Li-ion batteries exemplified by Mn-based cathodes – a critical review, *Energy Environ. Sci.*, 2018, 11(2), 243–257.
- 87 H. Zhang, I. Hasa and S. Passerini, Beyond Insertion for Na-Ion Batteries: Nanostructured Alloying and Conversion Anode Materials, *Adv. Energy Mater.*, 2018, 8(17), 1702582.
- 88 L. Wang, K. Liang, G. Wang and Y. Yang, Interface-engineered hematite nanocones as binder-free electrodes for high-performance lithium-ion batteries, *J. Mater. Chem. A*, 2018, 6(28), 13968–13974.
- 89 K. Liang, K. Marcus, S. Zhang, L. Zhou, Y. Li, S. T. De Oliveira, N. Orlovskaya, Y.-H. Sohn and Y. Yang, NiS_2/FeS Holey Film as Freestanding Electrode for High-Performance Lithium Battery, *Adv. Energy Mater.*, 2017, 7(22), 1701309.
- 90 K. Liang, K. Marcus, L. Guo, Z. Li, L. Zhou, Y. Li, S. T. De Oliveira, N. Orlovskaya, Y. H. Sohn and Y. Yang, A free-standing NiS_x porous film as a binder-free electrode for Mg-ion batteries, *Chem. Commun.*, 2017, 53(54), 7608–7611.
- 91 L. Shen, Q. Che, H. Li and X. Zhang, Mesoporous NiCo_2O_4 Nanowire Arrays Grown on Carbon Textiles as Binder-Free Flexible Electrodes for Energy Storage, *Adv. Funct. Mater.*, 2014, 24(18), 2630–2637.
- 92 F.-H. Du, B. Li, W. Fu, Y.-J. Xiong, K.-X. Wang and J.-S. Chen, Surface Binding of Polypyrrole on Porous Silicon Hollow Nanospheres for Li-Ion Battery Anodes

- with High Structure Stability, *Adv. Mater.*, 2014, **26**(35), 6145–6150.
- 93 Q. Lu, G. S. Hutchings, Y. Zhou, H. L. Xin, H. Zheng and F. Jiao, Nanostructured flexible Mg-modified LiMnPO_4 matrix as high-rate cathode materials for Li-ion batteries, *J. Mater. Chem. A*, 2014, **2**(18), 6368–6373.
 - 94 Y. Wang, Y. Wang, D. Jia, Z. Peng, Y. Xia and G. Zheng, All-Nanowire Based Li-Ion Full Cells Using Homologous Mn_2O_3 and LiMn_2O_4 , *Nano Lett.*, 2014, **14**(2), 1080–1084.
 - 95 Q. Wang, B. Mao, S. I. Stolarov and J. Sun, A review of lithium ion battery failure mechanisms and fire prevention strategies, *Prog. Energy Combust. Sci.*, 2019, **73**, 95–131.
 - 96 F. Han, J. Yue, C. Chen, N. Zhao, X. Fan, Z. Ma, T. Gao, F. Wang, X. Guo and C. Wang, Interphase Engineering Enabled All-Ceramic Lithium Battery, *Joule*, 2018, **2**(3), 497–508.
 - 97 X. Zhu, X. Jiang, X. Ai, H. Yang and Y. Cao, A Highly Thermostable Ceramic-Grafted Microporous Polyethylene Separator for Safer Lithium-Ion Batteries, *ACS Appl. Mater. Interfaces*, 2015, **7**(43), 24119–24126.
 - 98 M. He, X. Zhang, K. Jiang, J. Wang and Y. Wang, Pure Inorganic Separator for Lithium Ion Batteries, *ACS Appl. Mater. Interfaces*, 2015, **7**(1), 738–742.
 - 99 Q. Yang, Z. Zhang, X.-G. Sun, Y.-S. Hu, H. Xing and S. Dai, Ionic liquids and derived materials for lithium and sodium batteries, *Chem. Soc. Rev.*, 2018, **47**(6), 2020–2064.
 - 100 Q. Li and H. Ardebili, Flexible thin-film battery based on solid-like ionic liquid-polymer electrolyte, *J. Power Sources*, 2016, **303**, 17–21.
 - 101 M. Cheng, Y. Jiang, W. Yao, Y. Yuan, R. Deivanayagam, T. Foroozan, Z. Huang, B. Song, R. Rojaee, T. Shokuhfar, Y. Pan, J. Lu and R. Shahbazian-Yassar, Elevated-Temperature 3D Printing of Hybrid Solid-State Electrolyte for Li-Ion Batteries, *Adv. Mater.*, 2018, **30**(39), 1800615.
 - 102 Q. Ni, Y. Bai, Y. Li, L. Ling, L. Li, G. Chen, Z. Wang, H. Ren, F. Wu and C. Wu, 3D Electronic Channels Wrapped Large-Sized $\text{Na}_3\text{V}_2(\text{PO}_4)_3$ as Flexible Electrode for Sodium-Ion Batteries, *Small*, 2018, **14**(43), 1702864.
 - 103 A. Du, H. Zhang, Z. Zhang, J. Zhao, Z. Cui, Y. Zhao, S. Dong, L. Wang, X. Zhou and G. Cui, A Crosslinked Polytetrahydrofuran-Borate-Based Polymer Electrolyte Enabling Wide-Working-Temperature-Range Rechargeable Magnesium Batteries, *Adv. Mater.*, 2019, **31**(11), 1805930.
 - 104 G. Wang, Z. Yang, Y. Du and Y. Yang, Programmable Exposure of Pt Active Facets for Efficient Oxygen Reduction, *Angew. Chem., Int. Ed.*, 2019, **131**, 15995–16001.
 - 105 W. Niu, Z. Li, K. Marcus, L. Zhou, Y. Li, R. Ye, K. Liang and Y. Yang, Surface-Modified Porous Carbon Nitride Composites as Highly Efficient Electrocatalyst for Zn-Air Batteries, *Adv. Energy Mater.*, 2018, **8**(1), 1701642.
 - 106 W. Niu, S. Pakhira, K. Marcus, Z. Li, J. L. Mendoza-Cortes and Y. Yang, Apically Dominant Mechanism for Improving Catalytic Activities of N-Doped Carbon Nanotube Arrays in Rechargeable Zinc-Air Battery, *Adv. Energy Mater.*, 2018, **8**(20), 1800480.
 - 107 X. Han, X. Li, J. White, C. Zhong, Y. Deng, W. Hu and T. Ma, Metal-Air Batteries: From Static to Flow System, *Adv. Energy Mater.*, 2018, **8**(27), 1801396.
 - 108 Y. Li and J. Lu, Metal-Air Batteries: Will They Be the Future Electrochemical Energy Storage Device of Choice?, *ACS Energy Lett.*, 2017, **2**(6), 1370–1377.
 - 109 H.-F. Wang, C. Tang, B. Wang, B.-Q. Li, X. Cui and Q. Zhang, Defect-rich carbon fiber electrocatalysts with porous graphene skin for flexible solid-state zinc-air batteries, *Energy Storage Mater.*, 2018, **15**, 124–130.
 - 110 J. Jacas Biendicho, D. Noréus, C. Offer, G. Svensson, R. I. Smith and S. Hull, New Opportunities for Air Cathode Batteries; *in situ* Neutron Diffraction Measurements, *Front. Energy Res.*, 2018, **6**(69), 1–4.
 - 111 J. Pan, Y. Y. Xu, H. Yang, Z. Dong, H. Liu and B. Y. Xia, Advanced Architectures and Relatives of Air Electrodes in Zn-Air Batteries, *Adv. Sci.*, 2018, **5**(4), 1700691.
 - 112 D. Lee, H.-W. Kim, J.-M. Kim, K.-H. Kim and S.-Y. Lee, Flexible/Rechargeable Zn-Air Batteries Based on Multifunctional Heteronanomat Architecture, *ACS Appl. Mater. Interfaces*, 2018, **10**(26), 22210–22217.