

1 Enhancing ORR performance of bimetallic PdAg  
2 electrocatalysts by designing interactions between  
3 Pd and Ag

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12 Displacement, Cu UPD

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1 ABSTRACT: Precise tuning of the electronic properties of Ag/C using under potentially  
2 deposited (UPD) Cu and subsequent galvanic displacement to deposit atomically dispersed  
3 loading of Pd resulted in a robust bimetallic alloy with significant activity for the oxygen  
4 reduction reaction in alkaline media. The specific design of the catalyst and atomic arrangement  
5 of Pd-Ag outperforms conventional Pd/C and Ag/C commercial catalysts. The ORR activity of  
6 Pd deposited onto Ag/C was determined based on rotating disk electrode voltammetry studies,  
7 showing two-fold increase in Pd mass activities compared to Pd/C. While scanning transmission  
8 electron microscopy (STEM) coupled with electron energy loss spectroscopy (EELS) probed the  
9 uniformity of the nanoparticles, the origin of the outstanding activity was traced to the structural  
10 properties of the Pd-Ag interface as shown by X-ray absorption spectroscopy (XAS), along with  
11 X-ray photoelectron spectroscopy (XPS). Segregation of metals with a suitable geometric  
12 arrangement of Ag to Pd ratio at the interface, resulted in an increased performance where the  
13 active sites were key steps of oxygen bond breaking.

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1    **Introduction**

2    The oxygen reduction reaction (ORR) is an essential energy conversion process typically used for energy-  
3    storing devices such as batteries and fuel cells.<sup>1</sup> This is particularly true for fuel cell vehicles for which,  
4    the potential growth of the market and the concern for abundant gas emissions, urge for the design of  
5    catalytic systems with improved activity. Efforts are geared towards decreasing the amount of Pt or  
6    finding a substitute with adequate ORR activity.<sup>2-4</sup> A promising approach is to design new electrocatalysts  
7    having monolayer amounts of noble metal on a surface of suitable metallic nanoparticles.<sup>5-7</sup> For that  
8    reason, a catalyst support such as carbon is essential for the proper dispersal of metal nanoparticles by  
9    acting as a path of electrons, thus increasing electrical conductivity while simultaneously reducing metal  
10   loadings.<sup>8</sup> Electrochemical approaches generally lead to higher control over particle and distribution size,  
11   creating stronger interactions between metal and substrate.<sup>9</sup> Ag-based formulations have been found to  
12   have adequate Oxygen Reduction Reaction (ORR) activity, however the incorporated Pd yields additional  
13   benefits due to its unique catalytic features and distinct Ag-Pd interactions.<sup>10</sup> Experimental<sup>11</sup> and  
14   computational studies suggest that an enhanced Ag-Pd interface can improve significantly the catalytic  
15   activity of these materials.<sup>7, 12, 13</sup> In particular, the presence of Pd with Ag can lead to flexible control of  
16   the *d* band center via ligand effects exerted between the host metal and subsurface metal atoms.  
17   According to computational studies, a volcano plot shows that for ORR activity, Ag and Au bind weakly  
18   to oxygen, whereas other transition metals (e.g. Ni, Cu, Rh, Pd) bind strongly to both O<sub>2</sub> and hydroxyl  
19   species.<sup>12</sup> Therefore, a combination of Ag and Pd at a suitable geometric arrangement may give optimum  
20   binding of O for the ORR.<sup>14</sup>

21   The electrochemical oxygen reduction in alkaline media of Pd and Ag-based catalysts has been  
22   highlighted in recent review articles<sup>15-17</sup>. However, the electrochemical oxygen reduction behavior of Ag-  
23   containing alloys still requires further insight. The high reactivity for ORR over Pd-Ag alloy was recently  
24   described by the Stevenson group, where they produced stable bimetallic nanoalloys via co-reduction of  
25   Ag and Pd carboxylic acid complexes precursors at similar rates<sup>18</sup> in the presence of stabilizing ligands<sup>11</sup>.

1 In their study, Ag-Pd nanoalloys are shown to favor the oxygen reduction reaction at a Ag:Pd  
2 composition of 9:1. These alloys exhibited a suitable geometric arrangement allowing to work by an  
3 ensemble effect, where the Pd atoms can facilitate the binding of the initial oxygen and subsequently the  
4 Ag can help desorb the reaction products such as OH. In such a way, the full four-electron process takes  
5 place by combining the fast kinetics of Pd for the adsorption with rapid disproportionation on Ag for the  
6 desorption step. To the best of our knowledge, electrochemical routes, which provide an enrichment of  
7 Ag at a AgPd alloy, have not been previously reported.

8 Previous work by Lüsi et al. show the use of galvanic displacement on an electron beam-evaporated Cu  
9 used to prepare thin film Pd electrodes tested for the ORR reaction in alkaline media.<sup>19</sup> This active  
10 arrangement gave a 4-electron pathway similar to that of commercial Pd/C making Cu a suitable  
11 sacrificial agent to be galvanically displaced by Pd. For this work, Pd nanoparticles were systematically  
12 deposited onto an UPD layer of Cu-ad-atoms onto Ag/C via galvanic displacement, as well as Ag  
13 nanoparticles were deposited onto an UPD layer of Cu ad-atoms onto Pd/C.<sup>19</sup> Characteristics of the  
14 preparation method, properties and performance of the catalyst have been investigated in detail. The data  
15 suggest a correlation between the bimetallic interaction developed at nanoscale and the high catalytic  
16 activity observed. For consistency purposes, both Pd deposited onto Ag/C (Pd @ Ag/C) and Ag deposited  
17 onto Pd/C (Ag @ Pd/C) were prepared in tandem.

18 These bimetals were prepared using a high corrosion resistance cell made of a graphite plate, which acted  
19 as the working electrode, and a reducing potential was applied with respect to the Ag/AgCl reference  
20 electrode with a platinum flag used as auxiliary electrode. Commercial 30% Pd/C (E-TEK, 10 wt%,  
21 3.5nm Pd particle size) and 40% Ag/C (Premetek, 40 wt%, 5.0 nm Ag particle size) were used as  
22 reference catalysts. The details on sample preparation and the electrochemical parameters used are given  
23 in the Supporting Information (Figure S1), along with the detailed description of electrochemical testing  
24 conditions and electron microscopy characterization methods.

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1    **2. Experimental Section**

2    **2.1 Synthesis of bimets.** An electrochemical cell with a high corrosion resistance made of graphite  
3    acted as a working electrode (14 cm diameter). Two grams of commercially available carbon supported  
4    Pd nanoparticles were placed on the bottom of the cell. The Pd/C nanoparticles surface was  
5    electrochemically reduced by applying a potential between the working electrode and a leak-less Ag|AgCl  
6    electrode with a platinum flag as the auxiliary electrode. Afterwards, a concentrated CuSO<sub>4</sub> solution was  
7    added to adjust the concentration of Cu<sup>2+</sup> in the cell to 50 mM. A constant potential (100 mV vs Ag|AgCl)  
8    was then applied to form UPD Cu ad-atoms on the Pd nanoparticle surfaces. The solution was stirred  
9    while the constant potential was applied, to disperse the Pd/C particles in the electrolyte, ensuring the  
10   formation of UPD Cu monolayers on the Pd surfaces. Cu UPD is carried out until the current reached a  
11   steady current value near zero. After Cu monolayers, a Ag<sup>+</sup> solution of AgNO<sub>3</sub> in a 50 mM H<sub>2</sub>SO<sub>4</sub> on a  
12   separate reservoir was injected slowly into the cell while it remained vigorously stirring to allow Ag<sup>+</sup> ions  
13   to replace Cu monolayer on Pd surfaces via galvanic displacement. The amount of metal salt precursor  
14   used was calculated for a monolayer amount of catalyst with respect to the host metal while taking into  
15   consideration the stoichiometry of the reaction with respect to Cu<sup>+</sup>. The same technique was used to  
16   deposit Pd onto Ag/C using a solution of PdCl<sub>2</sub> with Pd<sup>2+</sup> as precursor afterwards.

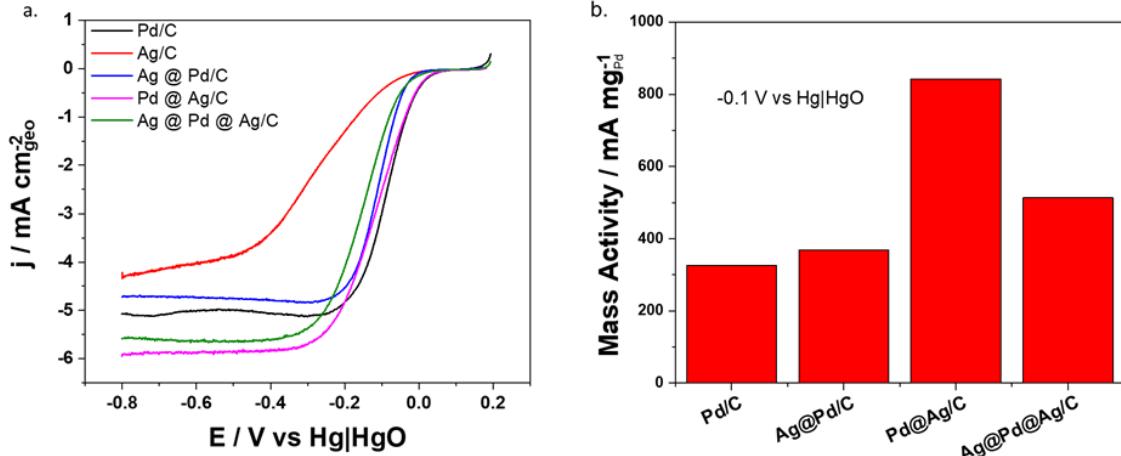
17    **2.2 Characterization Techniques** Transmission electron microscopy (TEM), high annular dark field  
18   scanning electron microscopy (HAADF-STEM), and electron energy loss spectroscopy (EELS) for  
19   elemental mapping were conducted on a high resolution, analytical scanning/transmission electron  
20   microscope (S/TEM, FEI Talos F200X). Images were taken operating at 200 keV at the Center for  
21   Functional Nanomaterials, BNL. XAS measurements were performed at the National Synchrotron Light  
22   Source (NSLS-II), Brookhaven National Laboratory (BNL using Beam Lines 8-ID and 8-BM. The  
23   measurements were carried out using the Pd L<sub>III</sub>-edge (3,173 keV), Pd K-edge (24,350 eV) and Ag K  
24   edge (25,514 eV) in fluorescence mode and a passivated implanter planar silicon (PIPS) detector. Data  
25   processing was performed using the IEFFIT package.<sup>20</sup> A commercial SPECS AP-XPS chamber equipped

1 with a PHOIBOS 150 EP MCD-9 analyzer at the Chemistry Division of BNL was used for XPS analysis.  
2 The C 1s (284.5 eV) feature was used for energy calibration.<sup>21</sup> The catalyst in form of a powder was  
3 pressed on an aluminum plate and then loaded into the XPS chamber. The mass weight % was determined  
4 by inductively coupled plasma-atomic emission spectroscopy (ICP-AES). The catalyst was  
5 electrochemically cycled from -0.8 to 0.2 V vs Hg|HgO at 20 mV/s for 10 cycles under Ar to clean the  
6 surface in 0.1M KOH. Following this cleaning procedure, an extended cyclic voltammetry (CV) from -  
7 0.9V to 0.9 V at 20 mV/s was performed to identify the oxidation/reduction peaks for both Ag and Pd.  
8 The electrolyte was then saturated with oxygen bubbling the solution for 10 minutes. ORR activity  
9 measurements were taken by rotating the electrode at 1600 rpm performing a linear sweep from 0.2V to -  
10 0.8V at 10 mV/s.

## 11 Results

12 The polarization curves at 1600 rpm for the studied catalysts for the ORR reaction together with their  
13 pure Pd and Ag metal components are shown in Figure 1a. Quantitative ORR kinetic parameters obtained  
14 at 1600 rpm under ORR conditions can be observed in Table 1. The overall performance of the samples  
15 was strongly influenced on the presence of Pd in the catalyst. Modifying Pd/C by incorporating Ag (blue  
16 curve in Figure 1a) decreases the diffusion limited current density ( $j_D$ ) and shifts the ORR half wave  
17 potential ( $E_{1/2}$ ) closer to Ag/C (red curve) thereby increasing its overpotential for the ORR as compared to  
18 Pd/C (black curve). Conversely, when modifying Ag/C with Pd, the ORR  $E_{1/2}$  was shifted to more  
19 positive values, towards Pd/C. In comparison to Ag/C the overpotential is significantly reduced along  
20 with an overall activity increment observed as an increase in  $j_D$  as can be observed in Figure 1a, Ag @  
21 Pd/C and Pd @ Ag/C outperformed Ag/C in terms of activity. This can be seen by the higher catalytic  
22 current density measured at -0.1V vs. Hg|HgO and the potential shift towards more positive potentials  
23 with values reported on Table 1. The Pd @ Ag/C sample was further modified by incorporating an  
24 additional layer of Ag via Cu UPD followed by galvanic displacement (Ag @ Pd @ Ag/C). As can be

1 observed from the ORR activity presented in Figure 1b, the Ag @ Pd @ Ag/C was less active than Pd @  
 2 Ag/C. Therefore, Pd @ Ag/C was further tested and compared to Ag @ Pd/C.



3  
 4 **Figure 1:** The electrochemical oxygen reduction reaction (ORR) activity comparison between Ag and Pd  
 5 alloys and their pure constituents: (a) linear sweep voltammogram from a rotating disk electrode  
 6 measurement at 1600 rpm in  $\text{O}_2$  saturated 0.1 M KOH from 0.2 V to -0.8 V vs Hg|HgO at 10 mV/s and  
 7 (b) bar plot representation of ORR activity at -0.1 V vs Hg|HgO normalized per Pd loading.

8 The plots used to obtain kinetic parameters of the ORR reaction are shown in Figure S2 as well as the  
 9 Koutechy Levich plots used to calculate the total number of electrons transferred (Figure S3). <sup>22</sup>For Pd @  
 10 Ag/C the net electron transfer was 4.0, close to what has been reported for pure Pd and also close to Pd/C  
 11 (3.9) as obtained in these measurements.<sup>11, 23</sup> The mass activity at -0.1 V vs Hg|HgO normalized by  
 12 nominal amount of Pd are shown in Figure 1 (b) for all the tested catalysts. Pd @ Ag/C is almost twice as  
 13 active than Pd/C based on the generated current density per Pd. Ag @ Pd/C and Ag @ Pd @ Ag/C were  
 14 found to be less active than Pd @ Ag/C. Changes in the oxygen reduction kinetic parameters indicate how  
 15 the introduction of small amounts of foreign metals onto the surface of a catalyst can greatly influences  
 16 the catalytic activity of the system. Mass weight % for the catalysts were determined by inductively  
 17 coupled plasma-atomic emission spectroscopy (ICP-AES) and the results are shown in Table S1.

1 The Pd @ Ag/C enhanced performance is also maintained during stability experiments where the  
 2 catalysts were cycled from 0.2 to -0.8 V vs Hg|HgO for 5000 potential cycles. Figure S4 shows that  
 3 incorporation of Pd onto the structure prolongs the usefulness of the catalysts with reductions in  $j_D$  of less  
 4 than 10%. Meanwhile, Ag/C reported a decrease in overall performance activity of 40%. These results  
 5 highlight the enhancement and overall performance of the catalysts due to the presence of Palladium.

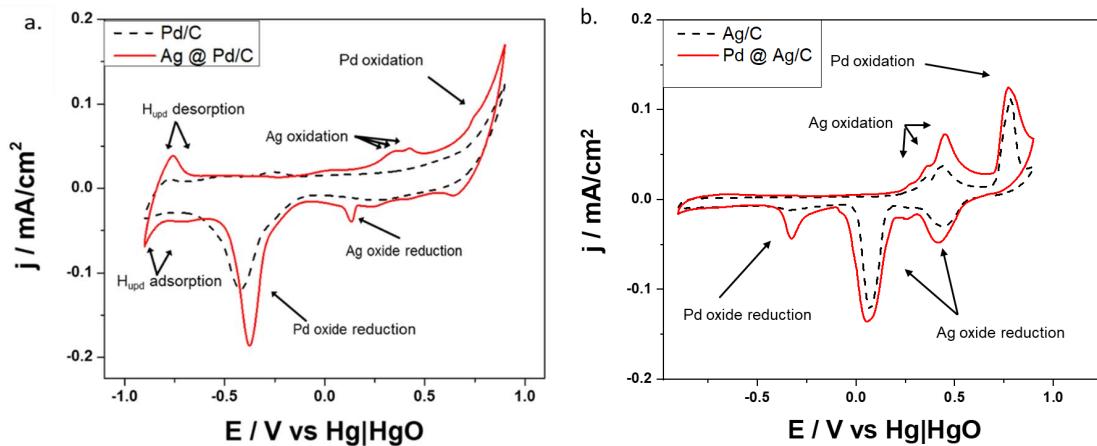
6 **Table 1:** Quantitative ORR kinetics parameters calculated for the ORR in 0.1M KOH

Catalyst	Normalized Pd mass activity (mA mg <sup>-1</sup> )	E <sub>onset</sub> (mV)	E <sub>1/2</sub> (mV)	No. of electrons*
Ag/C	-	-59	-261	2.5
Pd/C	341	-12	-86	3.7
Ag @ Pd/C	396	-6	-108	3.4
Pd @ Ag/C	878	-37	-117	4.0
Ag @ Pd @ Ag/C	578	-50	-156	3.4

7 \*Number of electrons were calculated applying the Koutechy-Levich equation

8 Cyclic voltammetry was used to probe the surface composition of the catalysts. Figure 2 presents the  
 9 cyclic voltammogram (CV) for the analyzed samples along with their starting materials (i.e. Pd/C and  
 10 Ag/C). The assignment of redox reactions to the respective peaks are indicated based on a comparison  
 11 with the data for pure Pd/C and Ag/C in Ar-saturated 0.1M KOH. All the peaks observed for the metals  
 12 prior to the electrochemical synthesis are also present on the catalyst, suggesting the presence of an alloy  
 13 where the surface is composed of both metals in the resulting structure. The CV for pure Pd/C contains  
 14 the respective characteristic adsorption and desorption of H<sub>upd</sub> at potentials of -0.6 to -1.0V, along with the  
 15 expected Pd oxide reduction at -0.40 V vs Hg|HgO. For the Ag/C, no distinctive adsorption and  
 16 desorption of H<sub>upd</sub> features are observed.<sup>24</sup> When the potential is scanned from 0.25 to 0.5 V vs Hg|HgO,  
 17 three characteristic oxidation peaks can be observed corresponding to Ag<sub>2</sub>O monolayer formation, AgOH  
 18 bulk and Ag<sub>2</sub>O bulk.<sup>25,26</sup> In the anodic sweep, the Ag @ Pd/C exhibits two characteristic peaks associated  
 19 to Ag and Pd oxidation at the surface while the cathodic sweep shows the reduction of the Ag<sub>2</sub>O and PdO,  
 20 indicating that both metals are electroactive at the surface of the catalyst. The voltammogram for Pd @

1 Ag/C also shows distinctive features corresponding to the presence of both metals at the surface. Pd @  
 2 Ag/C voltammetry profiles do not show distinctive hydrogen adsorption/desorption features like those for  
 3 Pd/C. This feature has been previously reported using Pd-Au alloys which hypothesized that a minimum  
 4 cluster size of Pd dimers is required to exhibit adsorption/desorption behavior.<sup>27</sup> Hence, this study  
 5 proposes that Pd atoms are dispersed in Ag rich surface domains, as the CV results demonstrate that both  
 6 studied metals are available at the surface, regardless of the deposition arrangement.

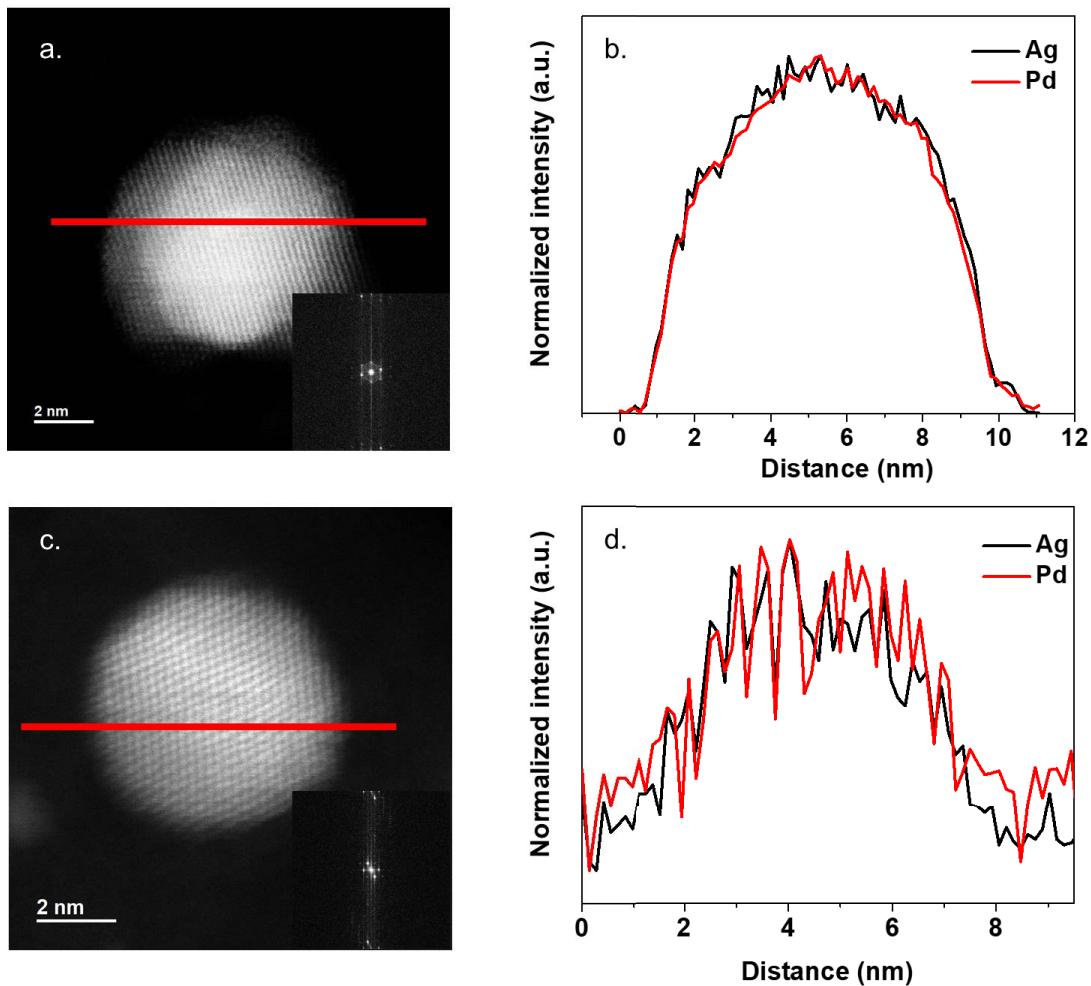


7 **Figure 2:** Cyclic voltammograms of (a) Pd/C and Ag@Pd/C, and (b) Ag/C and Pd@Ag/C measured in  
 8 Ar-purged 0.1 M KOH at 50 mV/s. The CVs in (a) show the characteristic peaks for (a) Pd/C and the  
 9 incorporation of Ag<sup>2+</sup>. The surface shows that the active surface is composed of both metals confirming  
 10 the behavior of an alloy. The CVs in (b) show Ag/C and the incorporation of Pd<sup>2+</sup>. The absence of H<sub>upd</sub>  
 11 peaks suggests minimum cluster sizes of Pd dimers are at the surface.

12 Adding a Ag layer onto Pd @ Ag/C still shows the features for Ag and Pd; however, the Ag peaks  
 13 become broader while the Pd reduction peak magnitude shifts (Figure S5). Thus, the CV validates that the  
 14 surface becomes Ag rich. In turn, this increased the overpotential for the ORR and was detrimental for the  
 15 activity as compared to the Pd @ Ag/C. It further indicates that the ORR reaction is strongly dependent  
 16 on the nature of the active sites present at the surface.

17 Details of the prepared catalyst's morphology were obtained by HR-TEM analysis on fresh samples.  
 18 Figure 3 (a) shows the image for Pd @ Ag/C where small Pd cover the Ag NP uniformly, while (b) shows  
 19 Ag being dispersed over Pd/C. The starting commercial Ag/C and Pd/C materials TEM images are shown  
 20

1 in Figure S6. The elemental composition, examined by EELS, for both alloys showed a homogeneous  
2 distribution of Pd and Ag particles and local probing revealed an average particle size of ~5 nm. These  
3 results confirm the behavior of the nanoparticle alloy consisting of both metals at the surface, as opposed  
4 to a core-shell. It has been previously reported that the composition of both metals at the surface can  
5 enhance the catalytic activity through an ensemble effect, where both metals interact in synergy at the  
6 reaction's active site promoting faster kinetics.<sup>11</sup>



7

8 **Figure 3:** (a) Scanning transmission electron microscopy (STEM) and (b) electron energy loss  
9 spectroscopy (EELS) of Pd @ Ag/C alloys created electrochemically and (c) STEM and (d) EELS of Ag  
10 @ Pd/C alloys created electrochemically. The uniform composition of the AgPd alloy is confirmed by the  
11 EELS line scan.

1 The increased performance for the bimetallic catalysts can be traced to a combination of factors of which  
2 ligand effects and the enrichment of metals at the surface of the catalyst seem to play a crucial role. For  
3 that matter, XAS measurements were taken at the L<sub>III</sub> (Pd) and K (Pd and Ag) absorption edges. XANES  
4 at the L<sub>III</sub>-edge is sensitive to the unoccupied portion of electronic density of state of *d* character. For Pd,  
5 the XANES data at the L<sub>III</sub> edge energy exhibits sharp, intense peak, which is an ideal probe for studies of  
6 Pd-based catalysts since the p-d transition is particularly sensitive to changes upon alloying.<sup>28</sup>

7 Figure S7 shows the resonance peak at the absorption Pd L<sub>III</sub> edge, also known as the white line (WL)  
8 arising from the 2p-4d transition.<sup>28</sup> The intensity of the WL decreases with *d*-band occupancy and this line  
9 is particularly strong for transition metals with a partially filled *d*-band. A distinct sharp white line was  
10 observed for the reference Pd(NO<sub>3</sub>)<sub>2</sub> and Pd/C materials. Both synthesized catalysts (Pd @ Ag/C and Ag  
11 @ Pd/C) exhibit an attenuation of the white line intensity, consistent with alloy formation.<sup>28</sup> Sham et. al  
12 studies found that relative to the pure element, Pd gains *d* and loses non-*d* (*s* and *p*) charge upon alloying.  
13 This phenomenon can be attributed to a decrease of the number of 4d holes in Pd mainly due to a shift of  
14 electron density from the Ag to the Pd metal based on differences in their electronegativity.<sup>29</sup> Not  
15 exposing the catalyst to the Cu UPD step before galvanic displacement (PdAg/C and AgPd/C in Figure  
16 S8) resulted in no intimate interaction between Ag and Pd. Figure S8 shows a comparison of the samples  
17 of PdAg/C and AgPd/C were their white line intensity is noticeably sharper and does not resemble that of  
18 the alloys obtained and the behavior obtained for Pd L<sub>III</sub> edge when alloyed. Hence, the electrochemical  
19 synthesis steps are crucial for an intimate Pd-Ag interaction.

20 The analysis of atomic structure of the alloys was carried out by EXAFS using both Pd and Ag K-edges  
21 to elucidate the active structure of the catalysts within nearest vicinity (the first coordination sphere) of  
22 each metal species. Figures 4(a-b) show the XANES spectra of Pd and Ag K-edges, and Figures 4 (c-d)  
23 show the Fourier transform magnitudes of the EXAFS data spectra of Pd and Ag K-edges presented in  
24 Figure S9. The Pd K-edge white line intensity for the alloy samples resemble that of Pd foil. On the other  
25 hand, the Pd/C sample has a sharper white line suggesting a higher oxidation state than pure Pd. For Ag

1 @ Pd/C, the electrochemical treatment modifies the Pd/C electronic structure and reduces Pd at the bulk  
2 with an electronic transfer from Ag to Pd. This coincides with the Fourier transform where the Pd foil  
3 magnitude is similar to those of the alloy. The Ag K-edge shows more remarkable differences for the  
4 alloyed samples as compared to the Ag foil presumably due to net electron transfers from Ag to Pd. The  
5 data strongly suggests a significant change in coordination for both elements as they interact with each  
6 other. For that matter, the results of EXAFS fitting as well as the fitting performed for each of the  
7 prepared catalyst at each respective metal's absorption edge, are summarized in Table 2. Since Pd and  
8 Ag are adjacent in the periodic table, their photoelectron scattering amplitudes and phase shifts are  
9 similar; the method could not, in principle discriminate between Pd-Pd and Pd-Ag bonds for Pd absorbers  
10 (or Ag-Ag and Ag-Pd for Ag absorbers). The subtle differences between neighboring species around the  
11 central atom makes it difficult to conclude whether an alloy is formed. However, their lattice parameters  
12 are significantly different and Ag has a larger atomic radius than Pd and the interaction between both  
13 metals change the bond length compared to their respective starting materials (Ag/C and Pd/C). From  
14 these results, it can be observed that the Pd-Pd coordination number (CN) for the foil is 12 with a bond  
15 length of 2.737 Å. However, the Pd/C nanoparticles used for the synthesis have a lower bond length,  
16 consistent with the smaller size of Pd nanoparticles as compared to the Pd foil bulk. Pd/C also exhibits a  
17 peak around 2.0 Å related to the Pd-O bond length as it has been previously reported.<sup>30</sup> When Ag is  
18 incorporated into the structure (Ag @ Pd/C), the bond length of Pd increases to 2.747 Å, which is longer  
19 than the initial Pd foil or the Pd/C radial distance and origins from the interaction of Pd with the Ag atom.  
20 From the Ag K-edge perspective, it can be observed that there is a decrease of bond length for Ag @  
21 Pd/C (from 2.860 to 2.786 Å).

22 On the other hand, for Pd @ Ag/C, the Pd and the Ag absorption edges show a similar trend in  
23 terms of the radial distance, obtained where the bond length of palladium increases and the bond length of  
24 Ag decreases. However, from the Pd edge perspective, the Pd-Pd CN value increased while from the Ag  
25 K-edge, the Ag-Ag(Pd) CN decreased to 1.7. This small CN is attributed to small interactions between

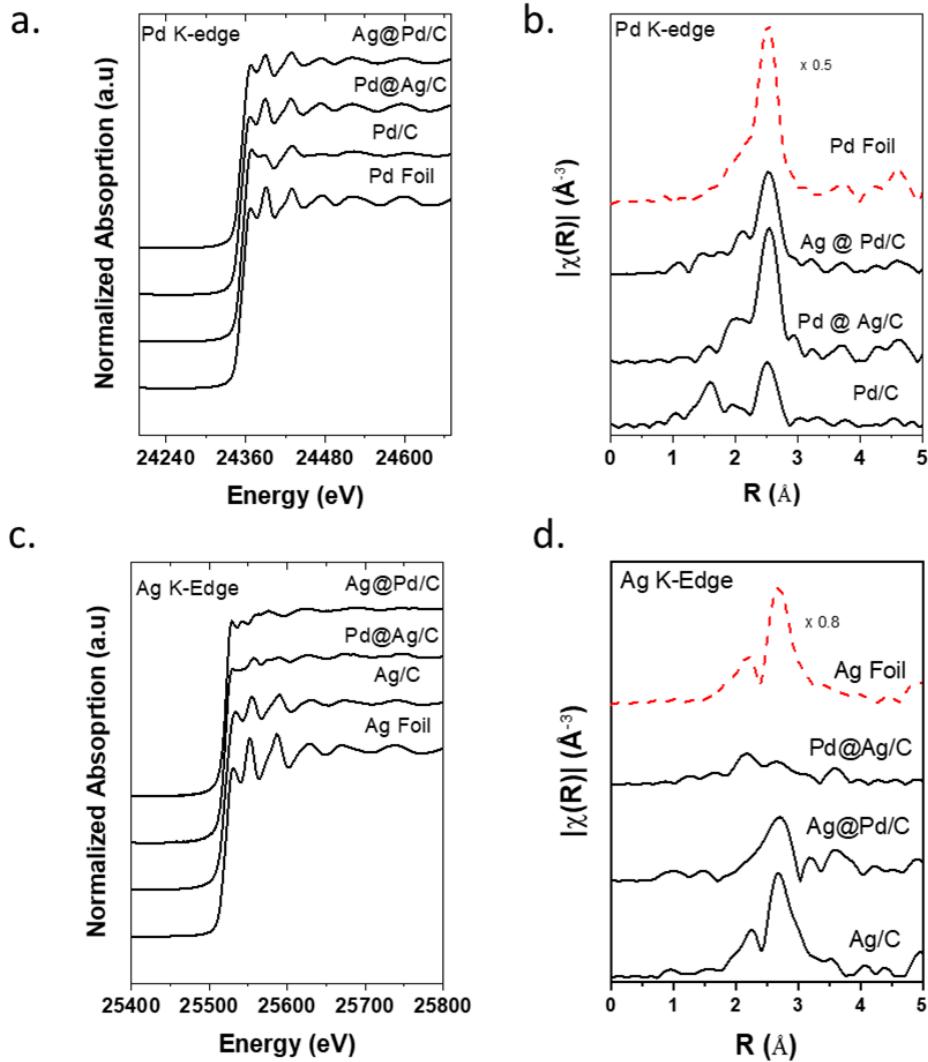
1 Ag-Cl from the  $\text{PdCl}_2$  precursor. For that matter, the synthesis was carried out using  $\text{Pd}(\text{NO}_3)_2$  as a  
2 precursor to substitute  $\text{PdCl}_2$ . Thereafter, the catalyst was evaluated for ORR activity (Figure S10) at 1600  
3 rpm, showing less catalytic improvement than using  $\text{PdCl}_2$  as precursor and was not used for further  
4 experiments.

5 XANES and EXAFS analysis confirms the strong interaction between Pd and Ag metals obtained using  
6 the electrochemical synthesis. There is a net electron transfer from Ag to Pd based on the reduced white  
7 line intensity observed in the XANES as well as the expansion of Pd and contraction of Ag radius for  
8 both alloys. However, electrochemical reactions are significantly more sensitive to the surface of the  
9 catalyst exposed to the solution. For that matter, XPS is significantly more sensitive to the surface  
10 composition rather than the bulk material, thus it was used to examine the chemical and electronic  
11 composition of Pd and Ag of these catalysts.

12 **Table 2:** Fit parameters of the catalysts determined by the EXAFS fitting

Sample Name	K-Edge	Near Neighbor Species	N	R (Å)	$\sigma^2$ (Å <sup>2</sup> )	$E_0$ (eV)
Ag foil	Ag	Ag	12	$2.860 \pm 0.002$	$0.0095 \pm 0.0003$	$0.6 \pm 0.3$
Ag/C	Ag	Ag	$10.7 \pm 0.6$	$2.861 \pm 0.003$	$0.0095 \pm 0.0005$	$0.8 \pm 0.4$
Ag @ Pd/C	Ag	Pd/Ag	$9.3 \pm 0.6$	$2.786 \pm 0.004$	$0.0096 \pm 0.0006$	$0.1 \pm 0.5$
Pd @ Ag/C	Ag	Cl	$1.7 \pm 0.6$	$2.72 \pm 0.03$	$0.001 \pm 0.005$	$5.6 \pm 2.7$
Pd foil	Pd	Pd	12	$2.737 \pm 0.002$	$0.0056 \pm 0.0002$	$-1.5 \pm 0.4$
Pd/C	Pd	O	$3.5 \pm 1.1$	$2.06 \pm 0.01$	$0.003 \pm 0.003$	$2.7 \pm 1.2$
		Pd	$5.5 \pm 1.0$	$2.732 \pm 0.007$	$0.007 \pm 0.001$	$2.7 \pm 1.2$
Ag @ Pd/C	Pd	O	$1.4 \pm 0.6$	$2.06 \pm 0.02$	$0.005 \pm 0.005$	$2.2 \pm 0.4$
		Pd/Ag	$8.7 \pm 0.5$	$2.747 \pm 0.002$	$0.0063 \pm 0.0003$	$2.2 \pm 0.4$
Pd @ Ag/C	Pd	Pd/Ag	10.3	$2.746 \pm 0.003$	$0.0065 \pm 0.0004$	$-2.6 \pm 0.5$

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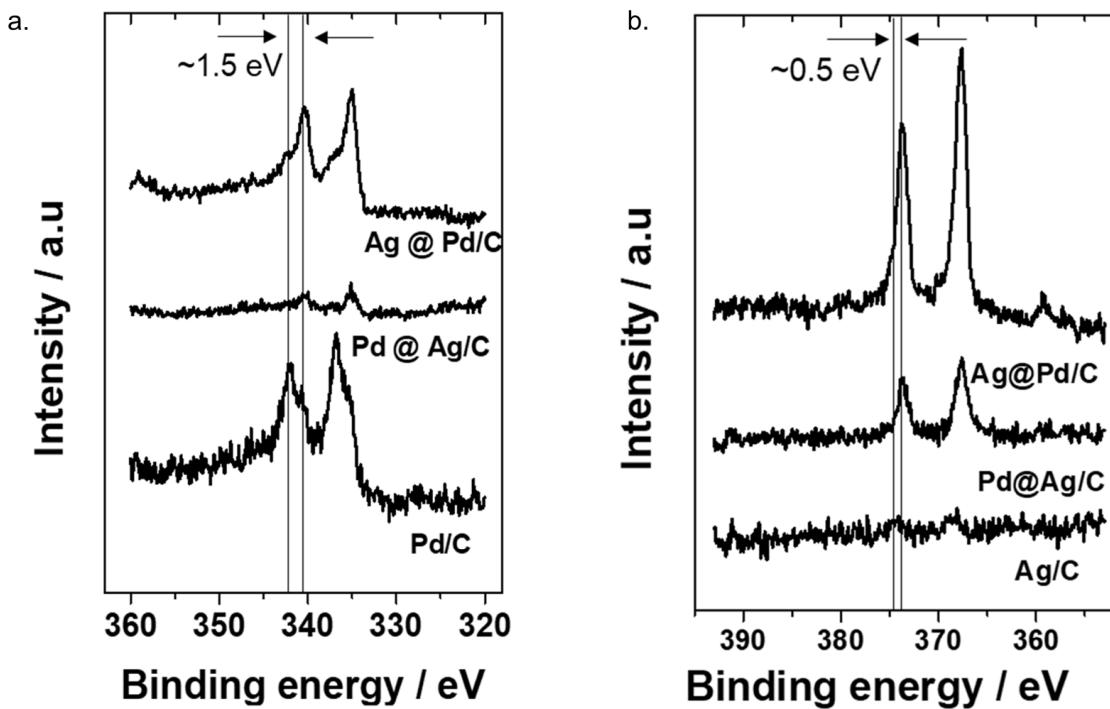
1  
2 **Figure 4:** (a) XANES data at the Pd K-edge and corresponding (b) Fourier transform magnitude. (c)  
3 XANES data at the Ag K absorption edge and (d) corresponding Fourier transform magnitude of the  
4 synthesized samples with corresponding reference.

5  
6 Pd/C spectra show a characteristic electronic transition ( $3d_{5/2}$ ) with a doublet located at 335.1 eV.<sup>31</sup> As  
7 demonstrated in Figure 5a, the Pd 3d spectrum for Pd/C with binding energy of 336.7 eV and 342.0 eV  
8 indicated that Pd was predominantly present in its oxidized state. Once the Ag is incorporated into the  
9 structure (Ag @ Pd/C), the binding energy shift to 335.1 and 340.3 eV, consistent with the observations  
10 from PdAg alloy formation at the surface. While Pd was expected to be enriched at the surface of Pd @

1 Ag/C based on the electrochemical technique used for deposition, the Pd 3d peak had a lower intensity  
2 compared to Ag @ Pd/C. This supports the hypothesis of preferential enrichment of Ag at the surface  
3 based on differences their surface energy.

4 Figure 5b exhibits the  $3d_{5/2}$  transition for Ag with a doublet located at 368.3 eV.<sup>31</sup> The initial Ag/C 3d  
5 binding energy peak is barely visible attributed to Ag being buried within the carbon which attenuates  
6 photoelectron traveling through an attenuating medium. Nonetheless, the Ag peak intensity increases  
7 when Pd is incorporated via galvanic displacement. This is an indication that the incorporation of Pd not  
8 only perturbed the electronic structure of Ag, but also exposed Ag to the surface. Ag @ Pd/C showed the  
9 sharpest peak mainly due to segregation effects from the enrichment of Ag at the surface. This behavior  
10 has been previously reported on a Pd-Ag film, where Pd gains electron density from Ag, shifting the  
11 binding energy of Ag due to a gain of charge density in the *d*-band, with a concomitant loss in the *sp*  
12 band.<sup>28, 32</sup> Consequently, Ag is enriching the surface to a higher extent in Ag @ Pd/C than in Pd @ Ag/C.  
13 By this interaction, the incorporation of Pd into Ag/C weakens the intensity of the  $3d_{5/2}$  transition signal  
14 compared to Ag incorporated into Pd/C.

15



1  
2 **Figure 5:** X-ray photoelectron spectroscopy spectra for (a) Pd 3d and (b) Ag 3d binding energy regions  
3 for both Pd:Ag alloys and the commercial catalysts Ag/C and Pd/C.  
4 The electronic shift according to XPS shows a synergy in activity between the alloy ratio relative to its  
5 pure components particularly for the Pd @ Ag/C catalyst with minimal loading. This synergy is favored  
6 by the high degree Pd-Ag versus Pd-Pd contacts, which modified the Pd electronic structure as evaluated  
7 by XPS and XAS. The electronic structure of Ag is similar to Au with a filled *d* shell, which explains why  
8 Ag lies on the weak binding site of the volcano plot for activity by *d* band charge theory. For that matter,  
9 Ag is expected to show weak oxygen binding resulting in low binding affinity towards the first electron  
10 transfer step. Increasing the *d* band charge density by incorporating Pd, facilitates the initial electron  
11 transfer which contributes to strong oxygen binding via ensemble effects where two different metal atoms  
12 catalyze distinct steps in the reaction mechanism.<sup>33</sup> This approach of combining metals with a resulting  
13 higher activity is based on simple thermodynamic principles assuming a simple mechanism where one  
14 metal breaks the oxygen bond of molecular O<sub>2</sub> and the other metal reduced the resulting chemisorbed  
15 atomic oxygen.<sup>33</sup>

1 Higher reactivity for the Pd @ Ag/C can be explained by the different metal composition at the surface of  
2 the catalysts compared to their compositions at the bulk. XAS data helped to elucidate the electronic  
3 distribution of the materials where they share a similar structure. There is a net transfer from the less  
4 electronegative Ag atom to Pd, ultimately reducing the white line intensity and resembling the Pd foil. On  
5 the other hand, the Ag spectra for the alloys exhibit a sharp white line in agreement to oxidized Ag for  
6 both alloys. However, these results do not account for the higher reactivity of one metal alloy over the  
7 other. As previously mentioned, the surface of the catalyst is strongly influenced by the segregation of  
8 metals to the surface. Although the bulk of the material may be composed of a specific ratio, the Ag alloy  
9 components is enriching the surface region, which is sensitive to electrochemical techniques. Nørksov *et*  
10 *al.* studied the chemical composition at the surface of an alloy and concluded that it may differ from the  
11 composition in the bulk.<sup>34</sup> It has been well noted that the surface composition of most alloys is  
12 particularly sensitive to the external conditions, including the methodology used for deposition. Although  
13 this hypothesis has been applied to numerous systems, the behavior obtained, does not necessarily predict  
14 the correct segregation under experimental conditions.

15 An XPS peak fitting for the elemental composition at the surface is shown in Figure S11. To obtain a  
16 comparable relative concentration, the areas were adjusted by the atomic sensitivity factor. For the Pd @  
17 Ag/C, atomic contribution at the surface was calculated and found a 1:2 ratio for Pd to Ag whereas the  
18 Ag@Pd/C the atomic ratio was found to be 1:1. For solids, surface segregation can be translated to the  
19 mixing of heat, surface energy difference and relative atomic size. Experimentally the surface energy has  
20 been calculated to be 1.05-1.38 J/m<sup>2</sup> for Pd and 1.38-2.17 J/m<sup>2</sup> for Ag, with ranges varying based on  
21 exposed facets at the surface. As a result, the surface energy of Ag is considerably higher than that of Pd  
22 while the lattice constant of bulk (4.09 Å) is larger than bulk Pd (3.88 Å).<sup>35</sup> Ag is most likely to be found  
23 at the surface since less strain results in less surface energy than those in the subsurface and core region  
24 leaving Ag more exposed.<sup>36</sup> Therefore, regardless of whether Ag is deposited onto Pd/C or Pd onto Ag/C,  
25 the surface becomes more Ag enriched. Interestingly, for the compositional ratio at the surface for the Pd

1 @ Ag/C has outstanding activity for the ORR, compared to the Ag @ Pd/C or Pd/C. An additional layer  
2 of Ag modifies the compositional ratio at the surface and results in a decrease in performance in activity.  
3 Having Ag and Pd at the surface deposited by electrochemical methods, increased the intimate mixing  
4 between both metals which led to an increase in both electronic and ensemble effects.

5

6 **Conclusion**

7 The design of bimetallic alloys with atomically precise noble metal loading is of great interest in the field  
8 of catalysis. Herein, we demonstrated an electrochemical method able to reduce a  $\text{Ag}^+$  or  $\text{Pd}^{2+}$  precursor  
9 onto a Pd/C and Ag/C, respectively, to form alloy nanoparticles with high activities for the oxygen  
10 reduction reaction (ORR) in alkaline media. The ORR mass activity relative to pure Pd were determined  
11 at -0.1V vs HgHgO showing mass activities of 321 mA/mg<sub>Pd</sub> for Pd, 820 mA/mg<sub>Pd</sub> for Pd@Ag/C, 366  
12 mA/mg<sub>Pd</sub> for Ag@Pd/C and 526 mA/mg<sub>Pd</sub> Ag@Pd @Ag/C. For the most active catalyst, the activity was  
13 increased twofold on a Pd mass basis by incorporating Pd onto Ag/C. Scanning transmission electron  
14 microscopy and electron energy loss spectroscopy probed the uniformity of the nanoparticle's local  
15 structure with no significant differences in terms of dispersion and size for each alloy. The improvement  
16 in electrocatalytic response was explained by a combination of factors involving ligand and geometric  
17 effects shown by XPS and ex-situ EXAFS analysis, respectively. The segregation of Ag on a rich Ag  
18 surface 2:1 for Pd@Ag/C greatly enhanced the ORR reaction kinetics and the number of electrons  
19 transferred with respect to a 1:1 Ag to Pd ratio on Ag@Pd/C alloy and pure Pd. Moreover, the  
20 enhancement in catalytic activity was associated to the presence of both Ag and Pd at the surface, where  
21 the steps of oxygen bond breaking and desorption for the ORR occur in synergy: the initial  $\text{O}_2$  adsorbs at  
22 the Pd atoms followed by the rapid disproportionation step on the Ag rich surface.

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14

15    **Supporting Information Available:** The following file is available free of charge. *SI Enhancing ORR*  
16    *performance of bimetallic PdAg electrocatalysts*. The following information is included in this document:  
17    Table of ICP-AES analysis of catalyst materials and figures of the synthesis of Pd/Ag/C catalysts by  
18    galvanic displacement of upd Cu, ORR RDE thermodynamic parameters of catalysts, catalysts ORR RDE  
19    measurements, stability ORR polarization curves of catalysts, cyclic voltammetry of catalysts, TEM and  
20    particle size histogram of catalysts, Pd L<sub>3</sub>-edge XANES of alloys with and without the Cu upd treatment  
21    relative to Pd/C and a Pd @ Ag/C, XAFS data in k space for Pd and Ag K-edges for all catalysts, ORR  
22    RDE measurements, X-ray photoelectron spectroscopy of Ag 3d and Pd 3d binding energy curve fitting.

23

24    **References**

- 25    1. Shao, M.; Chang, Q.; Dodelet, J.-P.; Chenitz, R., Recent Advances in Electrocatalysts for  
26    Oxygen Reduction Reaction. *Chem. Rev.* **2016**, *116* (6), 3594-3657.
- 27    2. Xiong, Y.; Yang, Y.; Feng, X.; DiSalvo, F. J.; Abruña, H. D., A Strategy for Increasing  
28    the Efficiency of the Oxygen Reduction Reaction in Mn-Doped Cobalt Ferrites. *J. Am. Chem.  
29    Soc.* **2019**, *141* (10), 4412-4421.
- 30    3. Zhong, H.; Luo, Y.; He, S.; Tang, P.; Li, D.; Alonso-Vante, N.; Feng, Y., Electrocatalytic  
31    Cobalt Nanoparticles Interacting with Nitrogen-Doped Carbon Nanotube in Situ Generated from

1 a Metal–Organic Framework for the Oxygen Reduction Reaction. *ACS Appl. Mater. Interfaces*  
2 **2017**, *9* (3), 2541-2549.

3 4. Chen, G.; Kuttiyiel, K. A.; Su, D.; Li, M.; Wang, C.-H.; Buceta, D.; Du, C.; Gao, Y.; Yin,  
4 G.; Sasaki, K.; Vukmirovic, M. B.; Adzic, R. R., Oxygen Reduction Kinetics on Pt Monolayer  
5 Shell Highly Affected by the Structure of Bimetallic AuNi Cores. *Chem. Mater.* **2016**, *28* (15),  
6 5274-5281.

7 5. Sasaki, K.; Naohara, H.; Cai, Y.; Choi, Y. M.; Liu, P.; Vukmirovic, M. B.; Wang, J. X.;  
8 Adzic, R. R., Core-Protected Platinum Monolayer Shell High-Stability Electrocatalysts for Fuel-  
9 Cell Cathodes. *Angew. Chem. Int. Ed.* **2010**, *49* (46), 8602-8607.

10 6. Sasaki, K.; Marinkovic, N.; Isaacs, H. S.; Adzic, R. R., Synchrotron-Based In Situ  
11 Characterization of Carbon-Supported Platinum and Platinum Monolayer Electrocatalysts. *ACS*  
12 *Catal.* **2016**, *6* (1), 69-76.

13 7. Zhang, J.; Lima, F. H. B.; Shao, M. H.; Sasaki, K.; Wang, J. X.; Hanson, J.; Adzic, R. R.,  
14 Platinum Monolayer on Non noble Metal–Noble Metal Core–Shell Nanoparticle Electrocatalysts  
15 for O<sub>2</sub> Reduction. *J. Phys. Chem. B* **2005**, *109* (48), 22701-22704.

16 8. Antolini, E., Formation of carbon-supported PtM alloys for low temperature fuel cells: a  
17 review. *Mater. Chem. Phys.* **2003**, *78* (3), 563-573.

18 9. Safavi, A.; Maleki, N.; Farjami, E., Electrodeposited Silver Nanoparticles on Carbon  
19 Ionic Liquid Electrode for Electrocatalytic Sensing of Hydrogen Peroxide. *Electroanal.* **2009**, *21*  
20 (13), 1533-1538.

21 10. Sekol, R. C.; Li, X.; Cohen, P.; Doubek, G.; Carmo, M.; Taylor, A. D., Silver palladium  
22 core–shell electrocatalyst supported on MWNTs for ORR in alkaline media. *Appl. Catal. B:*  
23 *Environ.* **2013**, *138-139*, 285-293.

24 11. Slanac, D. A.; Hardin, W. G.; Johnston, K. P.; Stevenson, K. J., Atomic Ensemble and  
25 Electronic Effects in Ag-Rich AgPd Nanoalloy Catalysts for Oxygen Reduction in Alkaline  
26 Media. *J. Am. Chem. Soc.* **2012**, *134* (23), 9812-9819.

27 12. Nørskov, J. K.; Rossmeisl, J.; Logadottir, A.; Lindqvist, L.; Kitchin, J. R.; Bligaard, T.;  
28 Jónsson, H., Origin of the Overpotential for Oxygen Reduction at a Fuel-Cell Cathode. *J. Phys.*  
29 *Chem. B* **2004**, *108* (46), 17886-17892.

30 13. Roy, A.; Pal, T., Silver-induced electronic drift in AgPd bimetallics: rationale for  
31 enhanced electrocatalytic activity of ethanol oxidation reaction. *New J. Chem.* **2017**, *41* (20),  
32 12278-12287.

33 14. Stamenkovic, V.; Mun, B. S.; Mayrhofer, K. J. J.; Ross, P. N.; Markovic, N. M.;  
34 Rossmeisl, J.; Greeley, J.; Nørskov, J. K., Changing the Activity of Electrocatalysts for Oxygen  
35 Reduction by Tuning the Surface Electronic Structure. *Angew. Chem. Int. Ed.* **2006**, *45* (18),  
36 2897-2901.

37 15. Erikson, H.; Sarapuu, A.; Solla-Gullón, J.; Tammeveski, K., Recent progress in oxygen  
38 reduction electrocatalysis on Pd-based catalysts. *J. Electroanal. Chem.* **2016**, *780*, 327-336.

39 16. Zhang, L.; Chang, Q.; Chen, H.; Shao, M., Recent advances in palladium-based  
40 electrocatalysts for fuel cell reactions and hydrogen evolution reaction. *Nano Energy* **2016**, *29*,  
41 198-219.

42 17. Erikson, H.; Sarapuu, A.; Tammeveski, K., Oxygen Reduction Reaction on Silver  
43 Catalysts in Alkaline Media: a Minireview. *ChemElectroChem.* **2019**, *6* (1), 73-86.

44 18. Yamamoto, M.; Kakiuchi, H.; Kashiwagi, Y.; Yoshida, Y.; Ohno, T.; Nakamoto, M.,  
45 Synthesis of Ag–Pd Alloy Nanoparticles Suitable as Precursors for Ionic Migration-Resistant  
46 Conductive Film. *Bull. Chem. Soc. Jpn* **2010**, *83* (11), 1386-1391.

1 19. Lüsi, M.; Erikson, H.; Merisalu, M.; Kasikov, A.; Matisen, L.; Sammelselg, V.;  
2 Tammeveski, K., Oxygen Electroreduction in Alkaline Solution on Pd Coatings Prepared by  
3 Galvanic Exchange of Copper. *Electrocatalysis* **2018**, 9 (3), 400-408.

4 20. Ravel, B.; Newville, M., ATHENA, ARTEMIS, HEPHAESTUS: data analysis for X-ray  
5 absorption spectroscopy using IFEFFIT. *J. Synchrotron Radiat.* **2005**, 12 (Pt 4), 537-41.

6 21. Blyth, R. I. R.; Buqa, H.; Netzer, F. P.; Ramsey, M. G.; Besenhard, J. O.; Golob, P.;  
7 Winter, M., XPS studies of graphite electrode materials for lithium ion batteries. *Appl. Surf. Sci.*  
8 **2000**, 167 (1-2), 99-106.

9 22. Bard, A. J.; Faulkner, L. R., *Electrochemical Methods: Fundamentals and Applications*.  
10 Wiley & Sons: New York, 2002.

11 23. Wang, M.; Qin, X.; Jiang, K.; Dong, Y.; Shao, M.; Cai, W.-B., Electrocatalytic Activities  
12 of Oxygen Reduction Reaction on Pd/C and Pd–B/C Catalysts. *J. Phys. Chem. C* **2017**, 121 (6),  
13 3416-3423.

14 24. Spendelow, J. S.; Wieckowski, A., Electrocatalysis of oxygen reduction and small  
15 alcohol oxidation in alkaline media. *Phys. Chem. Chem. Phys.* **2007**, 9 (21), 2654-2675.

16 25. Guo, J.; Hsu, A.; Chu, D.; Chen, R., Improving Oxygen Reduction Reaction Activities on  
17 Carbon-Supported Ag Nanoparticles in Alkaline Solutions. *J. Phys. Chem. C* **2010**, 114 (10),  
18 4324-4330.

19 26. Hepel, M.; Tomkiewicz, M., Study of the Initial Stages of Anodic Oxidation of  
20 Polycrystalline Silver in KOH Solutions. *J. Electrochem. Soc.* **1984**, 131 (6), 1288-1294.

21 27. Maroun, F.; Ozanam, F.; Magnussen, O. M.; Behm, R. J., The Role of Atomic Ensembles  
22 in the Reactivity of Bimetallic Electrocatalysts. *Science* **2001**, 293 (5536), 1811.

23 28. Coulthard, I.; Sham, T. K., Charge Redistribution in Pd-Ag Alloys from a Local  
24 Perspective. *Phys. Rev. Lett.* **1996**, 77 (23), 4824-4827.

25 29. C. Witjens, L.; Bitter, J. H.; van Dillen, A. J.; De Jong, K.; De Groot, F., *Pd L3 edge*  
26 *XANES investigation of the electronic and geometric structure of Pd/Ag-H membranes*. *Phys.*  
27 *Chem. Chem. Phys.* **2004**, 6, 3903-3906.

28 30. Sasaki, K.; Naohara, H.; Choi, Y.; Cai, Y.; Chen, W.-F.; Liu, P.; Adzic, R. R., Highly  
29 stable Pt monolayer on PdAu nanoparticle electrocatalysts for the oxygen reduction reaction.  
30 *Nat. Commun.* **2012**, 3, 1115.

31 31. Mouder, J. F.; Stickle, W. F.; Sobol, P. E.; Bomben, K. D., *Handbook of X-Ray*  
32 *Photoelectron Spectroscopy*. Perkin-Elmer-Corp: Eden Prairie, MN, 1992.

33 32. Sham, T. K.; Naftel, S. J.; Coulthard, I.,  $M_{3,2}$ -edge x-ray absorption near-edge structure  
34 spectroscopy: An alternative probe to the  $L_{3,2}$ -edge near-edge structure for the unoccupied  
35 densities of d states of 5d metals. *J. Appl. Phys.* **1996**, 79 (9), 7134-7138.

36 33. Fernández, J. L.; Walsh, D. A.; Bard, A. J., Thermodynamic Guidelines for the Design of  
37 Bimetallic Catalysts for Oxygen Electroreduction and Rapid Screening by Scanning  
38 Electrochemical Microscopy. M–Co (M: Pd, Ag, Au). *J. Am. Chem. Soc.* **2005**, 127 (1), 357-  
39 365.

40 34. Liu, P.; Norskov, J. K., Ligand and ensemble effects in adsorption on alloy surfaces.  
41 *Phys. Chem. Chem. Phys.* **2001**, 3 (17), 3814-3818.

42 35. Sinnott, S. B.; Stave, M. S.; Raeker, T. J.; DePristo, A. E., Corrected effective-medium  
43 study of metal-surface relaxation. *Phys. Rev. B* **1991**, 44 (16), 8927-8941.

44 36. Kuijers, F. J.; Ponec, V., The surface composition of PdAg alloys. *J. Catal.* **1979**, 60,  
45 100–109.