

## FRONT MATTER

### Title

Shape-Morphing Living Composites

### Authors

L. K. Rivera-Tarazona<sup>1</sup>, V. D. Bhat<sup>2</sup>, H. Kim<sup>1</sup>, Z. T. Campbell<sup>2</sup>, T. H. Ware<sup>1\*</sup>.

### Affiliations

<sup>1</sup>Department of Bioengineering, The University of Texas at Dallas, Richardson, Texas, USA.

<sup>2</sup>Department of Biological Sciences, The University of Texas at Dallas, Richardson, Texas, USA.

\*Correspondence to: taylor.ware@utdallas.edu

**One Sentence Summary:** Baker's yeast embedded in a hydrogel enables the engineering of smart materials that change shape in response to specific cues.

### Abstract

This work establishes a means to exploit genetic networks to create living-synthetic composites that change shape in response to specific biochemical or physical stimuli. Baker's yeast embedded in a hydrogel forms a responsive material where cellular proliferation leads to a controllable increase in the composite volume of up to 400%. Genetic manipulation of the yeast enables composites where volume change on exposure to L-histidine is 14× higher than volume change when exposed to D-histidine or other amino acids. By encoding an optogenetic switch into yeast, spatiotemporally-controlled shape change is induced with pulses of dim blue light (2.7 mW/cm<sup>2</sup>). These living, shape-changing materials may enable sensors or medical devices that respond to highly specific cues found within a biological milieu.

## MAIN TEXT

### Introduction

Materials that change shape enable mechanical activity in devices, such as smart garments, sensors, microfluidics, or drug-delivery platforms (1-4). In these devices, traditional actuators, like solenoids, are too large, heavy, or power intensive to be used. Shape change in synthetic polymers and gels can be triggered using temperature, electric fields, or chemicals (5-8). The specificity of the response is dictated, and limited, by the physical characteristics of the material (9). One approach to induce specificity in the physical characteristics of a hydrogel is to build polymer networks from biomacromolecules, such as DNA, allowing for detection of analytes that directly bind to these constituents (10). Binding of designed DNA sequences can induce 100-fold volumetric hydrogel expansion by successive extension of cross-links, using a DNA hybridization cascade. In the design of chemically-responsive hydrogels, this mechanism is limited to detection of analytes capable of highly specific binding motifs. In living organisms, direct DNA binding is not the typical mechanism by which sensing occurs. Genetic information in cells encodes

43 components that enable appropriate responses to a wide range of specific chemical and physical  
44 cues.

45 Composites that combine the tunable properties of synthetic materials and the responsive nature of  
46 living organisms represent a powerful strategy to imbue multifunctionality in a single material.  
47 Several living composites have been previously reported including self-healing concrete (11),  
48 ethanol-producing 3D-printed hydrogels (12), gels that self-heal using photosynthesis (13), and  
49 wearable fluorescent biosensors (14). However, these living composites lack the ability to respond  
50 mechanically to environmental cues. One example of a mechanically-active living composite is a  
51 bilayer of an elastomer and mammalian muscles that bends through contraction and relaxation of  
52 the muscle cells (15). However, muscle cells only thrive over a very narrow set of conditions,  
53 limiting the range of applications where these materials can be used. Notably, shape change in  
54 living organisms is not limited to contraction of muscles. Tissue morphogenesis in animals and  
55 plants is controlled in part by cellular proliferation (16, 17). However, a strategy that harnesses  
56 proliferation of living cells to control the shape change of synthetic materials has yet to be reported.

57 Here we describe hybrid materials where living *Saccharomyces cerevisiae* (i.e., Baker's Yeast or  
58 Brewer's Yeast) embedded within a polyacrylamide hydrogel proliferates in response to a  
59 combination of environmental cues, which induces shape change in the composite (Fig. 1A). By  
60 controlling cell loading or hydrogel stiffness, we control the magnitude of volume change in the  
61 composites. This shape change is further controlled by patterning proliferation within a monolith.  
62 Critically, yeast provide a versatile platform for genetic engineering of the conditions required for  
63 proliferation. Using this control, we design composites that respond only in the presence of a single  
64 chirality of a single amino acid or to brief pulses of dim visible light. We harness this shape change  
65 to create microfluidic channels that respond selectively to fluids flowing through the channel.

## 66 67 **Results and discussion**

68 *S. cerevisiae* is an ideal model organism to realize responsive, living composites. These unicellular  
69 organisms thrive within solid matrices (12), are much stiffer (1-10 MPa) (18) than many hydrogels  
70 (10-100 kPa) (19), and are known to survive over a wide range of conditions (20). Our key  
71 observation is that as these stiff cells proliferate within a solid hydrogel matrix, a global increase in  
72 volume is observed. We hypothesize that this volume increase is not due to ordinary swelling of a  
73 hydrogel, but instead is attributable to local displacement of the hydrogel by the newly formed cells.  
74 After the composite is exposed to the appropriate conditions for cell growth, a dramatic increase in  
75 cell count can be observed (Fig. 1B). To quantify the effect of proliferation on macroscopic volume  
76 change, living composites were polymerized with 0.9-1.1 billion cells/mL of pre-gel solution (6  
77 wt% of dry yeast). Composites were incubated in YPD (yeast extract, peptone, D-glucose) media  
78 at 30 °C for 48 h. YPD contains the necessary nutrients for the yeast, and as such, cell proliferation-  
79 induced shape change occurs, resulting in a change in area of  $124.2\% \pm 10.3\%$  and a volume change  
80 of  $200.9\% \pm 2.4\%$  (Fig. 1C, movie S1). By incubating composites in media without a fermentable  
81 carbon source (D-glucose), the area of the disk only increased by  $6.3\% \pm 0.4\%$  as the yeast are  
82 incapable of proliferation (Fig. 1D, fig. S1). Similarly, hydrogels without encapsulated yeast  
83 incubated in YPD only undergo a volume change of  $1.2\% \pm 0.5\%$  (fig. S2). The shape change of  
84 the living composites in rich media is also not attributable to passive cell size changes; cell viability  
85 is required for shape change to occur. We pattern cell viability, using UV light (254 nm) exposed  
86 through a mask, in living composites covalently bound to glass (Fig. 1E). Only the regions of the  
87 hydrogel not exposed to UV, the letters "ALIVE," contain viable cells and undergo a volume  
88 increase on exposure to YPD. This expansion is greater than 110% of the initial film thickness after  
89 36 h (Fig. 1F, fig. S3). Shape change is accompanied by a change in topography, from smooth to  
90 rough, as the growing colonies deform the surface in a heterogeneous manner at the sub-mm scale.

The described experiments show that yeast proliferation is the primary mechanism associated with volume change in these hybrid living materials.

Proliferation-driven shape change can be tuned by controlling the initial composition of the living composite (Fig. 2A, 2B). On varying the initial concentration of yeast from 1 wt% to 18 wt%, the volume change after 48 h increases from  $123.8\% \pm 3.9\%$  to  $337.2\% \pm 17.4\%$ . This volume change is accompanied by a concomitant increase in dry mass, which varies from  $177\% \pm 11\%$  to  $320\% \pm 35\%$  (Fig. 2A). While we observe that some cells escape from the composite and proliferate in the media, this increase in dry mass suggests that most of the cells are retained in the hydrogel matrix. We also note that the shape of the grown composites is largely stable for over 128 days in deionized water at room temperature (fig. S4). The increase in dry mass and shape stability further supports our hypothesis that CO<sub>2</sub> production or passive swelling from the hydrogel matrix is not the mechanism responsible for shape change. This mass change represents new material that can be produced on demand with only as much external intervention or equipment as is needed to ferment grape juice. In the case of materials with 18 wt% yeast, the solid components of the as-synthesized composites are 35.9% polymer and 64.1% yeast. After growth, the yeast content increases to 85.6%. These growing composites may provide opportunities to produce materials directly from renewable feedstocks or even waste streams (21).

The mechanical properties of the hydrogel matrix also control the proliferation-induced shape change. By altering the feed ratio of crosslinker from 0.05 wt/v% to 0.6 wt/v%, at constant yeast loading (6 wt% dry yeast) and acrylamide concentration (10 wt/v%), the Young's modulus of the composites after synthesis increases from  $8 \text{ kPa} \pm 1 \text{ kPa}$  to  $204 \text{ kPa} \pm 16 \text{ kPa}$ . As stiffness increases, the volume change during cell proliferation decreases from  $255.8\% \pm 7.3\%$  to  $107.9\% \pm 1.2\%$  (Fig. 2B, fig. S5). We attribute this decrease to increased elastic resistance to the expanding colonies, perhaps resulting in limited cell proliferation. Given the tradeoffs between composite stiffness, yeast loading, and volume change, we selected composites with 0.1 wt/v% crosslinker and 6 wt% yeast for further studies as these composites have relatively high initial elastic modulus and large stimulus-response (fig. S6).

Spatial control of volume change can be programmed to yield composites that morph controllably from 2D to 3D. Informed by prior work where spatially-controlled swelling is used to guide shape selection in hydrogels (22-26), we fabricated composite disks (12 mm diameter and 0.5 mm thickness) and used UV light (254 nm) to kill cells in programmed areas (insets of Fig. 2C, 2D). After irradiation, no shape change is observed when the hydrogel is equilibrated in water, indicating that passive swelling of the gel is not significantly altered. After incubation in YPD, spatially-controlled proliferation induces a 2D to 3D transformation. The flat disk shown in Figure 2C grows in area in the center of the disk, while being constrained around the perimeter, resulting in a hemispherical cap (+ Gaussian curvature). By contrast, the disk depicted in Figure 2D grows along the perimeter, while being constrained in the center, resulting in a saddle-like geometry (– Gaussian curvature).

Programming of the stimulus that induces shape change of living composites can be achieved by genetic manipulation of the yeast. *S. cerevisiae* is a model eukaryote commonly used for heterologous protein expression (27-29). The yeast strain we employ (L40) is deficient in L-histidine metabolism. This metabolic feature, termed auxotrophy, prevents proliferation in the absence of L-histidine in the growth environment (Fig. 3A). This strain was used to fabricate composites that morph into 3D helical shapes only in the presence of L-histidine. Rectangular free-standing films were patterned with UV light to cause cell death in the areas indicated in Fig. 3B. Incubation for 48 h in selective media lacking L-histidine did not affect the shape of the composite. When these composites were incubated in media containing L-histidine, the flat films morph into a helix (Fig. 3C). Over 48 h in media lacking L-histidine, disks of these composites only increase in



control composites undergo a volume change of over 260% in both light and dark, and negative control composites grow only  $103.1\% \pm 20.1\%$  in the light and  $10.2\% \pm 5.3\%$  in the dark (Fig. 4E, fig. S9). We note that traditional photoresponsive polymers, which use light to power shape change, typically require irradiation intensities of more than  $100 \text{ mW/cm}^2$  (33, 34). By comparison, these living composites rely on light to trigger an optogenetic switch that has been optimized by evolution. The activation of this switch subsequently enables a metabolically-powered change in volume. As a result of this pathway, the time-averaged intensity required is at least  $2250\times$  smaller than traditional photoresponsive polymers.

The combination of patterned cell viability and patterned light illumination can be used to provide spatiotemporal control of complex shape change in living composites comprised of transformed yeast with the optogenetic switch. The cell viability within a film of the living composite was patterned using UV light to leave only two circular regions on the film with viable cells (fig. S10). As shown in Figure 2C, proliferation should lead to the formation of a hemispherical cap. By exposing these two regions sequentially, each region sequentially actuates from flat to hemispherical (Fig. 4F).

Living composites undergo cell proliferation-induced shape change controlled by the initial composition of the composite or by patterning regions of viable cells. These materials capitalize genetic control of biological mechanisms, namely cellular proliferation, to enable responsiveness in topography or shape in response to specific cues. A host of devices from drug delivery platforms to environmental sensors could be enabled by these findings.

## Materials and Methods

### Materials

Acrylamide, n,n'-methylenebisacrylamide (MBAA), ammonium persulfate (APS), n,n,n',n'-tetramethylethylenediamine (TEMED), 3-amino-1,2,4-triazole (3-AT), L-histidine, adenine sulfate, sulforhodamine B, bisphenol A ethoxylate diacrylate (BPA), and poly(ethylene glycol) diacrylate (PEG-DA) (700 g/mol) were purchased from Sigma-Aldrich. The photoinitiator Irgacure 369 (I-369) was donated by BASF corporation. Methacryloxyethyl thiocarbonyl Rhodamine B (PolyFluor 570) was purchased from Polysciences. Commercial yeast (*Saccharomyces cerevisiae*, Active Dry yeast, Fleischmann's) was purchased from Tom Thumb (Richardson, TX). Yeast extract, yeast nitrogen base without amino acids, peptone, D-(+)-glucose, D-histidine, and trypan Blue were purchased from Fisher Scientific. 3-(Trimethoxysilyl) propyl methacrylate (TPM) was purchased from Acros Organics. Rain-X® was purchased from Wal-Mart (Richardson, TX). All chemicals were used as received without further purification.

### Genetically-engineered yeast strains and plasmids

The genotype of the L40 yeast strain is *MATa ade2 his3 leu2 trp1 LYS::lexA-HIS3 URA3::lexA-LacZ* (ATCC, MYA-3332) (35). L40 yeast were transformed with experimental constructs CRY2 LexA DNA-binding fusion in the expression vector pDBTrp (pDBTrp-LexABD-CRY2FL) (Plasmid #78210, Addgene) and a separate CIB1 Gal4 activation domain fusion vector pGADT7 (pGal4AD-CIB1) (Plasmid #28245, Addgene). pDBTrp-LexABD-CRY2FL along with pGADT7 empty vector was used as negative control. In the positive control, we made use of a previously described interaction between *Caenorhabditis elegans* FBF2 (residues 121-C-terminus fused to the Gal4 activation domain present in pGADT7) and CPB1 (residues 1-80 fused to the LexA DNA-binding domain encoded by the pBTM116) (32,36).

## Mold construction

For volume change and mechanical testing experiments, molds were made of two glass slides (75x51mm) previously cleaned with Rain-X® to avoid gel adhesion. Slides were separated with 1 mm or 500 µm rectangular spacers, wrapped with parafilm closing one of the open sides, and fixed using binder clips.

For living composite coatings, molds (75x25mm) with one glass slide cleaned with Rain-X® and one treated with a methacrylate-functionalized silane were assembled. The two glass slides were separated with two 250µm polystyrene spacers on each side and fixed with binder clips. For the silane treatment, glass slides were cleaned following a similar process described in the literature (37). Briefly, glass slides were sonicated for 5 min in acetone and isopropanol mixtures and rinsed three times in dH<sub>2</sub>O. Afterwards, substrates were sonicated for 30 min in a mixture of water and Alconox cleaner (Alconox Inc. USA), rinsed, and stored in dH<sub>2</sub>O overnight. For silanization, glass slides were modified for 30 min with a 5 v/v% mixture of TPM in toluene at 65°C. Then, the slides were rinsed with toluene, dried with N<sub>2</sub> gas, and baked on a hotplate at 120°C for 5 min.

## Determination of cell density

To determine cell concentrations in active dried yeast, cell density was measured with a UV/Vis spectrophotometer by observing optical density at 660 nm. Briefly, 50 mL mixture with 0.6 g yeast in dH<sub>2</sub>O were prepared. Then, a 1:10 dilution was made by mixing 0.1 mL of the mixture with 0.9 mL of dH<sub>2</sub>O. Diluted samples were pipetted into a 1 mL cuvette for spectrophotometer measurements. Optical densities between 1 and 1.1 were measured, which correspond to numbers of cells of  $1.89 \times 10^7$  and  $2.25 \times 10^7$  cells respectively. These results indicate that the active dried yeast contained between 15 and 18 billion cells per gram.

## Preparation of living-composite materials with active dried yeast

Polyacrylamide hydrogels with embedded yeast were prepared at room temperature by free radical polymerization of acrylamide monomer and MBAA crosslinker. Stock solutions of 0.4 g/mL acrylamide and 0.02 g/mL MBAA were prepared in dH<sub>2</sub>O to create polyacrylamide gel precursor solutions. All pre-gel solutions were prepared with a final concentration of 10 wt/v% acrylamide. Pre-gel solutions were prepared with a final concentration of 0.1 wt/v% MBAA and ~ 1 billion cells/mL of pre-gel solution, unless otherwise noted. To polymerize these solutions, a 10 wt/v% APS stock solution was added at 1% of the total solution volume, and TEMED was added at a ratio of 0.1% of total solution volume. Polymerizing solutions were then vortexed for 3 s and quickly pipetted into molds. Filled molds were flipped every 45 s while polymerization occurred to avoid yeast settling. After 10 min, polymerized living composites were de-molded and rinsed three times with dH<sub>2</sub>O to remove unpolymerized acrylamide residues. Living composites were stored in dH<sub>2</sub>O for 24 h prior to mechanical testing and volume change experiments. For mechanical testing and volume change experiments, pre-gel solutions were prepared with 0.05, 0.1, 0.3 and 0.6 wt/v% MBAA and 6 wt/v% yeast (~ 1 billion cells/mL of pre-gel solution). To test volume change and Young's modulus with varying yeast content, composites with final concentrations of 0, 1, 6, 12, and 18 wt% yeast were prepared.

## Area, volume and mass change quantification of living-composites embedding active dried yeast

Living composites embedded with commercial active dried yeast were cut into 10 mm diameter disks. The dimensions of each disk were measured prior to incubation. Samples with varying dry

yeast, and crosslinker concentration were incubated at 30°C in YPD rich media without agitation (1% yeast extract, 2% peptone, and 2% D-glucose). To measure area and volume changes, three samples for each composition were incubated in 7 mL of rich media that was changed every 6 h. Area change was measured every hour for 48 h using a MightyScope 5M digital microscope, and volume changes were measured every 24 h for 48 h using a Canon Rebel T5i camera.

Dry mass change was obtained by weighing samples with varying yeast content before and after cell proliferation. Briefly, one set of living composites that was not exposed to media was dried at 30°C under vacuum to allow water evaporation. An identical set of living composites was incubated in YPD media for 48 h with media change every 6 h and then dried under the same conditions. Upon drying, samples were weighed, and mass was measured. Data presented are an average of 3 samples per composition.

### Material characterization

Samples (3 mm x 3 mm x 1 mm) were cut from polymerized living composites with varying yeast and crosslinker content after equilibration in dH<sub>2</sub>O. Compression testing was performed using a MicroSquisher (CellScale biomaterials testing). Briefly, a tungsten beam 1.016 mm in diameter was glued to a 6 mm x 6 mm platen on one end. This compliant beam was attached to an actuator with a cantilever beam grip at the opposite end from the platen. Samples were loaded to the test chamber filled with dH<sub>2</sub>O at room temperature. The beam was brought into contact with the sample and then moved at a rate of 0.5 mm/min. Force as a function of displacement was measured along the height (1 mm dimension) of the samples and calculated by the Microsquisher software using the beam's stiffness, displacement, and length. Strains from 1 to 10% were used to calculate Young's modulus, as the stress-strain response in this region was linear.

### Optical images of living composites

Microscopic imaging was carried out using an Olympus FV3000RS confocal laser scanning microscope. To visualize embedded yeast cell budding, living composites with 10 wt/v% acrylamide and 0.1 wt/v% MBAA were synthesized mixing approximately  $1 \times 10^6$  cells/mL of the pre-gel solution. Before polymerization, a 0.05 wt/v% aqueous solution of PolyFluor 570 was added at 1% of the total pre-gel solution volume. Cells, after incubation for 48 h in YPD media, were further stained by submerging samples in a 0.05 wt/v% aqueous solution of Trypan Blue for 3 min and then washed two times in dH<sub>2</sub>O. Budding of embedded cells and new colonies were observed throughout the thickness and area of the imaged samples (n=3).

### Macroscopic fluorescence images

For fluorescence imaging, living composites were dyed with a 0.05 wt/v% solution of sulforhodamine B in water. By shining light at a wavelength of 455 nm, fluorescent images of the UV patterned coatings and free-standing structures were obtained using a DSLR camera (Canon Rebel T5i) fitted with a red filter (Hoya HMC R25A). This filter blocks light below 600 nm, thus allowing visualization of the emitted light.

### UV photopatterning of composites with active dried yeast

Living composites covalently bound to methacrylate-functionalized glass molds with 500  $\mu$ m thickness were prepared as described above. Composites were allowed to equilibrate in water before UV exposure. A shadow mask of the word "ALIVE" was designed in AutoCAD and laser cut from

black polymer sheets. Irradiation with 254 nm UV light with an intensity of 2 mW/cm<sup>2</sup> was performed from one side for 35 min using an UVP UVLink 1000 crosslinker chamber. Samples were placed on a dark background during irradiation.

For UV patterning of free-standing composites, 12 mm diameter disks were cut from films 500 μm thick and patterned to induce cell death in a 6 mm diameter inner circle or a 3 mm wide ring pattern, using aluminum foil as a mask. Irradiation with 254 nm UV light with an intensity of 2 mW/cm<sup>2</sup> was performed from one side for 35 min using the UV chamber.

For cylindrical helix patterning, living composite samples were synthesized with the same composition of monomers as described above, and 0.9-1.1 billion cells/mL of pre-gel solution. Samples were cut into rectangular shapes (40 mm length, 5 mm height and 0.5 mm thickness) and patterned to induce cell death in 2.3 mm wide rectangles separated by 1.1 mm and positioned at 56° angle along the length of the samples (Fig. 3A). Irradiation was performed with the same wavelength, intensity, and time as described above.

After irradiation, living composites bound to glass were incubated for 36 h at 30 °C with a media change every 12 h. Free-standing disks were incubated for 48 h at 30 °C with media change every 12 h. These samples were then imaged using a fluorescent dye as described above. Cylindrical helix films were incubated for 48 h at room temperature with media change every 6 h and then imaged using a MightyScope 5M digital microscope every 5 min.

#### Topography measurements of UV photopatterned living composite coatings

Topography (Fig. 1F) of living composite coatings was imaged using a digital microscope (Keyence VHX-1000). To characterize the change in film thickness after cell proliferation, measurements along the depth profile of the letter “A” were taken. The camera limit points were set by focusing on the highest point of the grown letter and on the coated UV-killed surface. Between these limits, pictures were taken at 100× magnification along the surface of the letter. Images were then stitched using the Keyence software.

#### Quantifying shape change in living composites with auxotrophic yeast strain

Prior to composite synthesis, the auxotrophic yeast strain (CRY with empty vector, denoted as negative control strain) was grown overnight in selective media (0.7% yeast nitrogen base w/o amino acids, 2% D-glucose, and appropriate amino acid supplements (38) lacking tryptophan, leucine and histidine) containing L-histidine. Subsequently, overgrowths were made in 50 mL of YPAD (1% yeast extract, 2% peptone, 0.004% adenine sulfate, 2% D-glucose) media at 30°C for 15 h. Growth was followed by measuring optical density at 660 nm until desired yeast concentration was reached (OD<sub>660</sub> = 1-1.1). Cells were then centrifuged in 50 mL conical tubes and washed twice in distilled water before encapsulation. Composites with encapsulated auxotrophic yeast were synthesized by using 10 wt/v% acrylamide and 0.1 wt/v% MBAA with 0.9-1.1 billion cells/mL of pre-gel solution, as described above. For volume change experiments, living composites were first equilibrated in water for 24 h and then cut into 10 mm diameter disks with a thickness of 500 μm. Disks were incubated in selective media lacking L-histidine with 10 mM 3AT (HIS3 gene competitive inhibitor) for 48 h at 30°C, with a media change every 12 h. After this time, composites were incubated in selective media containing L-histidine and 10mM 3AT for another 48 h, with a media change every 12 h. Disks were measured after growth for quantification of volume change.



Identical disks were also exposed to selective media containing L-histidine, selective media containing D-histidine, a stereoisomer of the natural amino acid L-Histidine, and selective media without L-histidine. These, composites were incubated for 72 h at 30°C with a media change every 12 h. Data presented are an average of 3 samples per experiment.

#### Controlled blockage of microfluidic device

For shape change experiments using a microfluidic device, composites were cast into microfluidic polymer micromolds. These molds were built with 75x51mm glass slides. One of the two slides was functionalized with methacrylate groups by the processes described above. The other slide was coated in Rain-X®. Slides were separated with one 250 µm polystyrene spacer on each side and BPA mixture with 1 wt% I-369 photoinitiator was pipetted into the mold. Using a Vivitek D912HD (B9Creator) projector with the UV filter removed and the optics modified to decrease the focal length, a positive mold of a microfluidic device with an inlet channel 600µm wide and four outlet channels 400µm wide was polymerized onto the methacrylate-functionalized glass slide. To remove all unpolymerized BPA, slides were cleaned by multiple washes between acetone and isopropanol. This micromold was then used to create a cell for the polymerization of the living composite. After polymerization, microfluidic devices were allowed to equilibrate in dH<sub>2</sub>O for 24 h. Using UV-patterning (254 nm wavelength, 35 min, 2 mW/cm<sup>2</sup> intensity) most of the composite was rendered inviable. Two of the four outlet channels, as indicated in Fig. 3F, were kept alive by preventing UV exposure on the channel areas with a shadow mask. Selective media containing L-histidine was then flowed at a rate of 34 µL/min through one set of microfluidic devices (n=3) and selective media without L-histidine was flowed through another set of devices (n=3) at the same rate. Before and after media flow, channels were injected with a PEG-DA solution mixed with sulforhodamine B aqueous solution at 1% of the total PEG-DA volume. Flowing this solution allowed for visualization of the flow through the microchannels before and after growth. Topographical quantification of these samples was performed following the same process described for living composite coatings.

#### Yeast two-hybrid assays

L40 cells were co-transformed with the appropriate plasmids as described above and, in the literature (39). Cells were plated on selective media agar plates with 10 mM 3AT. Two replicate plates were grown in the dark and two were irradiated with blue light at 455 nm for 2 s every 2 min. Plates were incubated at 30°C for 3 days

Transformants were grown to saturation in selective media lacking L-histidine overnight at 30°C. Afterward, aliquots (100 µL) of the saturated cultures were outgrown in 1 mL of fresh minimal media and incubated at 30°C for 4 h. Optical density at 660 nm was recorded using a Spark® 20M multimode reader. β-Galactosidase expression was quantified using the Beta-Glo® Assay System (Promega). Briefly, 50 µL of cell cultures were added to an equal volume of reagent and allowed to incubate for 45 min prior to quantification. The luminescence values were normalized to cell densities for each culture.

#### Optogenetic control of shape change

Control of proliferation by exposure to blue light is achieved by encapsulating experimental yeast strains that express photosensory proteins. Pre-gel solutions with a composition of 10 wt/v% acrylamide and 0.1 wt/v% crosslinker were mixed with 0.9-1.1 billion transformed cells/mL of

solution. Positive and negative control strains were encapsulated using the same pre-gel composition and cell concentration. For volume change experiments, composites with each of the three strains were cut into 10 mm diameter disks with 500  $\mu\text{m}$  thickness. Samples were incubated in selective media lacking L-histidine with 10 mM 3AT for 72 h with media change every 12 h. Then, samples were irradiated with blue light (455 nm) (Fig. 4C) or kept in the dark. The intensity of irradiation was measured with a solar power meter (Amprobe Solar-100) and set at 2.7 mW/cm<sup>2</sup>. For blue light experiments, samples were irradiated for 2 s every 2 min. Volume changes were measured every 24 h for all samples before and after incubation. Data presented are an average of 3 samples per experiment.

For spatiotemporal control of proliferation, free-standing films with encapsulated experimental yeast strain were UV patterned as shown in fig. S10. Films were kept in dH<sub>2</sub>O for 24 h in the dark before blue light exposure. Films were then incubated in selective media with 10 mM 3AT at 30°C in the dark for 72 h. For the first 48 h, only the left half of the film is exposed to blue light to induce proliferation of cells. After this time, the right half was irradiated for 24 h. Images are representative from 3 trials.

## References and Notes

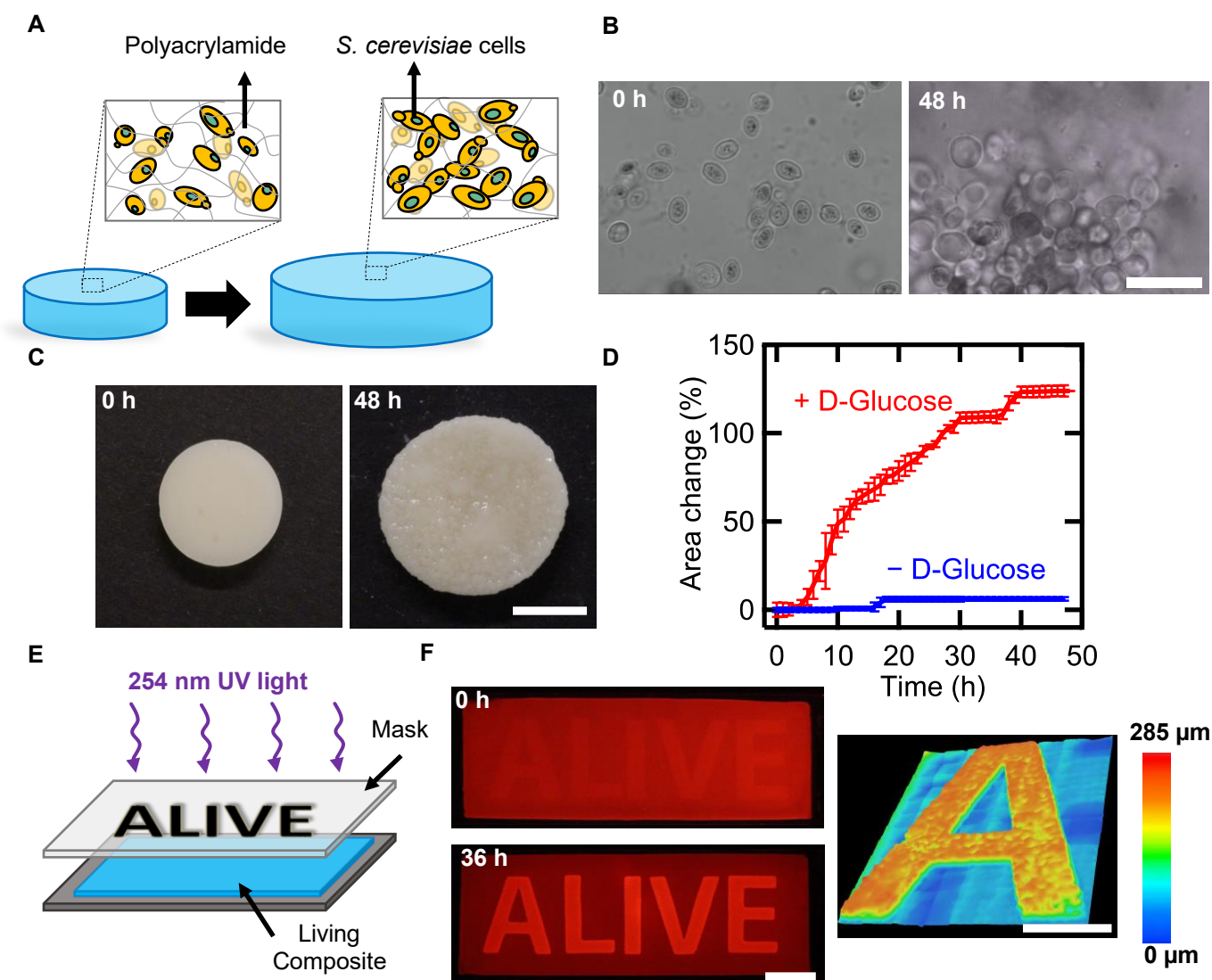
1. W. Wang *et al.*, Harnessing the hygroscopic and biofluorescent behaviors of genetically tractable microbial cells to design biohybrid wearables. *Science advances* **3**, e1601984 (2017).
2. W. Xu, K. S. Kwok, D. H. Gracias, Ultrathin Shape Change Smart Materials. *Accounts of chemical research* **51**, 436-444 (2018).
3. W. Hilber, Stimulus-active polymer actuators for next-generation microfluidic devices. *Applied Physics A* **122**, 751 (2016).
4. C. S. Haines *et al.*, Artificial muscles from fishing line and sewing thread. *science* **343**, 868-872 (2014).
5. R. Pelrine, R. Kornbluh, G. Kofod, High-strain actuator materials based on dielectric elastomers. *Advanced Materials* **12**, 1223-1225 (2000).
6. D. L. Thomsen *et al.*, Liquid crystal elastomers with mechanical properties of a muscle. *Macromolecules* **34**, 5868-5875 (2001).
7. T. Chung, A. Romo-Uribe, P. T. Mather, Two-way reversible shape memory in a semicrystalline network. *Macromolecules* **41**, 184-192 (2008).
8. S.-J. Jeon, A. W. Hauser, R. C. Hayward, Shape-morphing materials from stimuli-responsive hydrogel hybrids. *Accounts of chemical research* **50**, 161-169 (2017).
9. A. Lendlein, S. Kelch, Shape-memory polymers. *Angewandte Chemie International Edition* **41**, 2034-2057 (2002).
10. A. Cangialosi *et al.*, DNA sequence-directed shape change of photopatterned hydrogels via high-degree swelling. *Science* **357**, 1126-1130 (2017).
11. H. M. Jonkers, Bacteria-based self-healing concrete. *Heron*, *56* (1/2), (2011).
12. A. Saha *et al.*, Additive Manufacturing of Catalytically Active Living Materials. *ACS applied materials & interfaces* **10**, 13373-13380 (2018).
13. S. Y. Kwak *et al.*, Polymethacrylamide and Carbon Composites that Grow, Strengthen, and Self-Repair using Ambient Carbon Dioxide Fixation. *Advanced Materials* **30**, 1804037 (2018).
14. X. Liu *et al.*, Stretchable living materials and devices with hydrogel-elastomer hybrids hosting programmed cells. *Proceedings of the National Academy of Sciences* **114**, 2200-2205 (2017).

15. A. W. Feinberg *et al.*, Muscular thin films for building actuators and powering devices. *Science* **317**, 1366-1370 (2007).
16. A. J. Hughes *et al.*, Engineered tissue folding by mechanical compaction of the mesenchyme. *Developmental cell* **44**, 165-178. e166 (2018).
17. S. J. Streichan, C. R. Hoerner, T. Schneidt, D. Holzer, L. Hufnagel, Spatial constraints control cell proliferation in tissues. *Proceedings of the National Academy of Sciences* **111**, 5586-5591 (2014).
18. N. Minc, A. Boudaoud, F. Chang, Mechanical forces of fission yeast growth. *Current Biology* **19**, 1096-1101 (2009).
19. P. Calvert, Hydrogels for soft machines. *Advanced materials* **21**, 743-756 (2009).
20. X. Zhao, F. Bai, Mechanisms of yeast stress tolerance and its manipulation for efficient fuel ethanol production. *Journal of biotechnology* **144**, 23-30 (2009).
21. Y.-Q. Tang *et al.*, Ethanol production from kitchen waste using the flocculating yeast *Saccharomyces cerevisiae* strain KF-7. *Biomass and Bioenergy* **32**, 1037-1045 (2008).
22. Y. Klein, E. Efrati, E. Sharon, Shaping of elastic sheets by prescription of non-Euclidean metrics. *Science* **315**, 1116-1120 (2007).
23. A. S. Gladman, E. A. Matsumoto, R. G. Nuzzo, L. Mahadevan, J. A. Lewis, Biomimetic 4D printing. *Nature materials* **15**, 413 (2016).
24. E. Sharon, E. Efrati, The mechanics of non-Euclidean plates. *Soft Matter* **6**, 5693-5704 (2010).
25. A. Nojoomi, H. Arslan, K. Lee, K. Yum, Bioinspired 3D structures with programmable morphologies and motions. *Nature communications* **9**, 3705 (2018).
26. J. Kim, J. A. Hanna, M. Byun, C. D. Santangelo, R. C. Hayward, Designing responsive buckled surfaces by halftone gel lithography. *Science* **335**, 1201-1205 (2012).
27. S. Fields, O.-k. Song, A novel genetic system to detect protein-protein interactions. *Nature* **340**, 245 (1989).
28. M. A. Romanos, C. A. Scorer, J. J. Clare, Foreign gene expression in yeast: a review. *Yeast* **8**, 423-488 (1992).
29. A. P. Bollon, *Recombinant DNA products: Insulin, interferon and growth hormone*. (CRC Press, 2017).
30. X. Hu, Q. An, G. Li, S. Tao, J. Liu, Imprinted photonic polymers for chiral recognition. *Angewandte Chemie* **118**, 8325-8328 (2006).
31. M. J. Kennedy *et al.*, Rapid blue-light-mediated induction of protein interactions in living cells. *Nature methods* **7**, 973 (2010).
32. Z. T. Campbell *et al.*, Identification of a conserved interface between PUF and CPEB proteins. *Journal of Biological Chemistry* **287**, 18854-18862 (2012).
33. M. Yamada *et al.*, Photomobile polymer materials: towards light-driven plastic motors. *Angewandte Chemie* **120**, 5064-5066 (2008).
34. C. L. Van Oosten, C. W. Bastiaansen, D. J. Broer, Printed artificial cilia from liquid-crystal network actuators modularly driven by light. *Nature materials* **8**, 677 (2009).
35. A. B. Vojtek, S. M. Hollenberg, J. A. Cooper, Mammalian Ras interacts directly with the serine/threonine kinase raf. *Cell*. **74**, 205-214 (1993).
36. C. Luitjens *et al.*, CPEB proteins control two key steps in spermatogenesis in *C. elegans*. *Genes and Development*. **14**, 2596-2609 (2000).
37. D. Karnaushenko *et al.*, Biomimetic microelectronics for regenerative neuronal cuff implants. *Advanced Materials* **27**, 6797-6805 (2015).
38. W. Xiao, *Yeast Protocols*. (Humana Press, Totowa, NJ, 2006).
39. R. D. Gietz, R. A. Woods, in *Methods in enzymology*. (Elsevier, 2002), vol. 350, pp. 87-96.

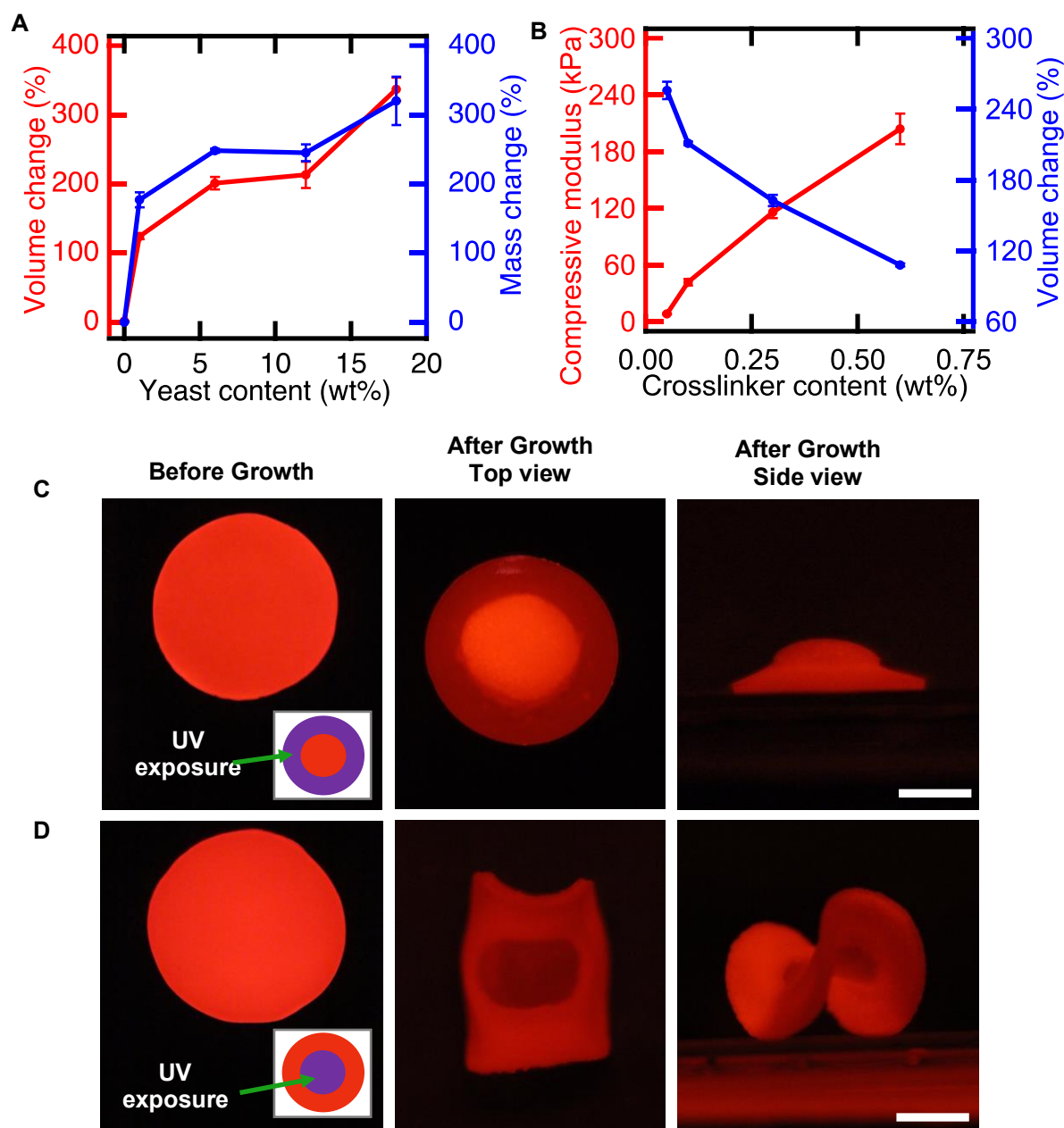
## Acknowledgments

NIH grants R01NS100788 (ZTC). This material is partially based upon work supported by the National Science Foundation under Grant Nos. 1752846 and 1663367. **Author contributions:** THW and ZTC conceptualized, provided supervision of, and acquired funding for the research. LKRT, VDB, and HK investigated. LKRT wrote the original draft. All authors reviewed and edited the manuscript. **Competing interests:** Authors declare no competing interests. **Data and materials availability:** All data is available in the main text or the supplementary materials.

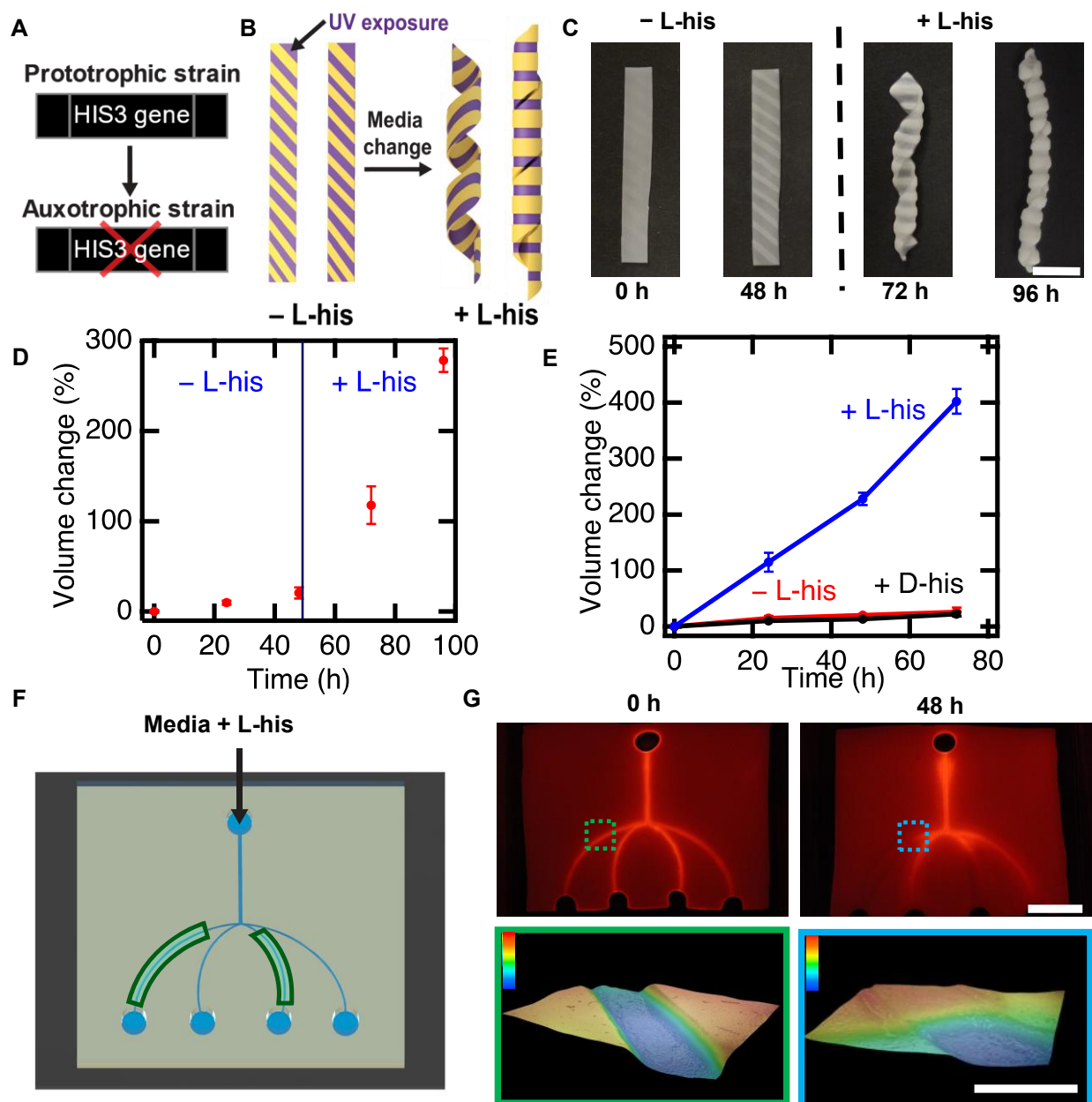
# Figures



**Fig. 1. Controlled expansion of polyacrylamide gels by proliferation of yeast.** (A) Schematic of shape-change in living composites. In YPD, yeast proliferate and cause expansion in the polymer matrix. (B) Optical micrographs of a living composite before and after growth in media (Scale bar: 30  $\mu\text{m}$ ). (C) Macroscopic expansion of a living composite gel with 6 wt% yeast (Scale bar: 7 mm). (D) Area change over time of a sample with 6 wt% yeast in the presence of media with and without glucose. (E) Photopatterning process of a living composite. (F) Fluorescence images of a living composite after UV patterning (top) and after incubation in YPD (bottom) (Scale bar: 10 mm). Topography of an initially flat living composite after exposure to YPD (right) (Scale bar: 5 mm). Each data point represents the mean ( $n=3$ ), and error bars represent standard deviation. Trend lines are only intended to guide the eye.

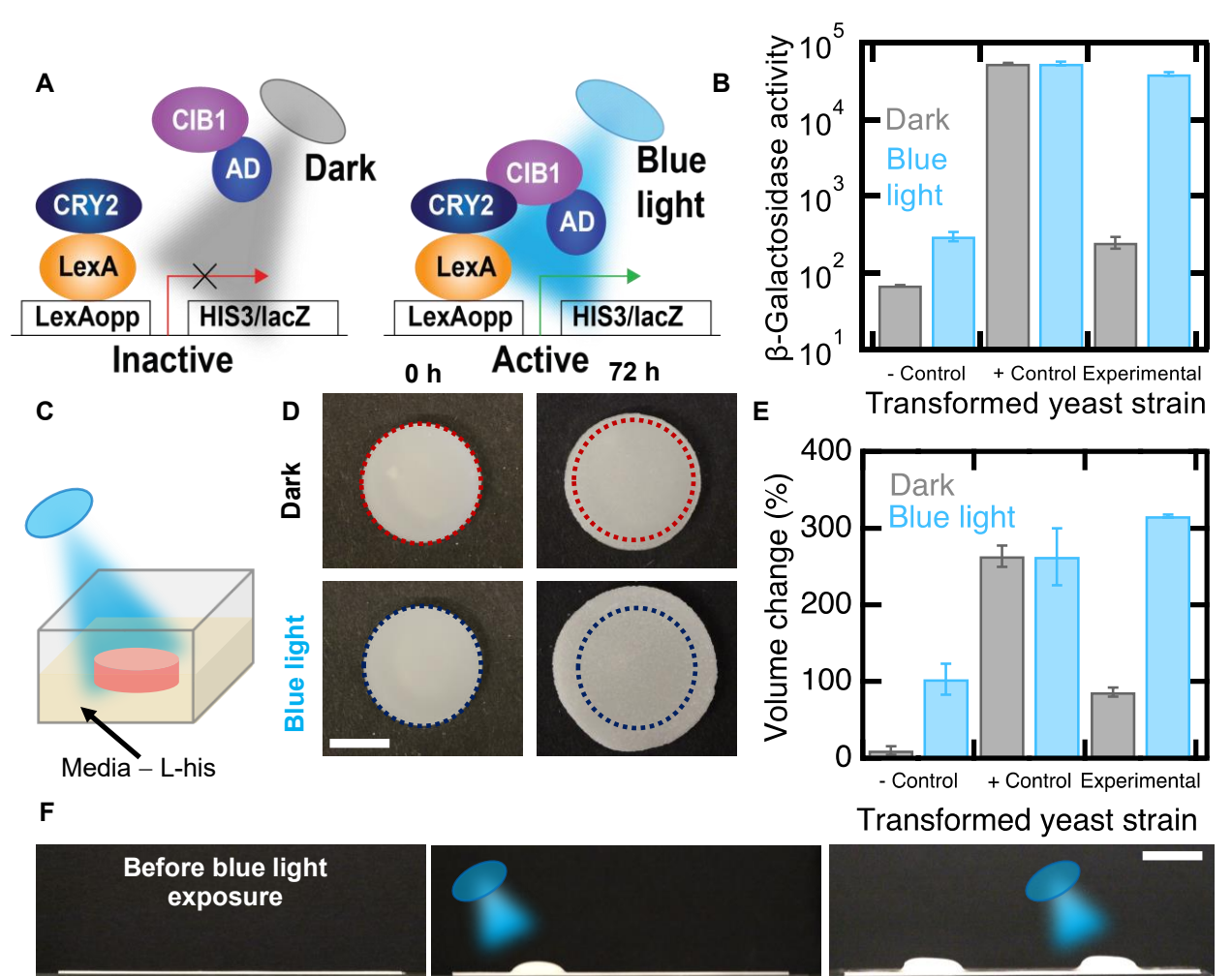


**Fig. 2. Shape change of living composites can be controlled.** (A) Volume and mass change of living composites as a function of yeast content. (B) Compressive modulus and volume change as a function of crosslinker content. (C) A flat disk exposed to spatially-patterned UV light (left) in a 3 mm wide ring pattern (inset). After incubation in media, a hat-like structure with positive Gaussian curvature is observed (center, right). (Scale bar: 5 mm). (D) A flat disk exposed to spatially-patterned UV light (left) in a 6 mm inner circle (inset). Upon incubation in media, a saddle-like structure with negative Gaussian curvature is observed (center, right). (Scale bar: 5 mm). Each data point represents the mean ( $n=3$ ), and error bars represent standard deviation. Trend lines are only intended to guide the eye.



**Fig. 3. Genetic engineering enables controlled composite response to specific cues.** (A) Deletion of the HIS3 gene results in failure to proliferate in media lacking histidine. (B) Schematic of a UV-patterned living composite with growth triggered by the amino acid L-histidine. (C) UV-patterned living composites with auxotrophic yeast do not substantially change in shape in media lacking L-histidine. Shape change into a helical structure after incubation in media containing L-histidine (Scale bar: 10 mm) (D) Volume change over time for auxotrophic living composites before and after L-histidine exposure. (E) Volume change over time for auxotrophic living composites incubated in media lacking histidine, with D-histidine, or with L-histidine. (F) Schematic of a living microfluidic device where the composites forming the channels indicated in green contain living auxotrophic yeast. (G) Fluorescence image of fluid traversing the microfluidic device before exposure to media (top left) (Scale bar 10 mm). Fluorescence image of fluid traversing the microfluidic device after media containing L-histidine flows for 48 h through the channels. Topography of a living channel before and after (color scale: 0-0.3 mm) growth (bottom) (Scale bar: 1 mm). Each data point represents the mean (n=3), and error bars represent standard deviation. Trend lines are only intended to guide the eye.





**Fig. 4. Genetic engineering enables optogenetic control of shape change.** (A) Schematic of a light sensitive yeast two-hybrid. Blue light induces expression of HIS3 and *lacZ* reporters by inducing conformational changes in CRY2 to favor interaction with CIB1. Reporter genes are transcribed by recruitment of the Gal4 activation domain (AD). (B)  $\beta$ -Galactosidase assays of an auxotrophic strain lacking CIB1 (negative control), a strain not auxotrophic for L-histidine in the dark (positive control), and the auxotrophic strain depicted in A (experimental). (C) Schematic of a living composite irradiated with blue light in growth media lacking L-histidine. (D) Volume change of living composites with experimental yeast irradiated with blue light or kept in the dark (Scale bars: 5 mm). (E) Volume change of living composites with each yeast strain when exposed to blue light or kept in the dark. (F) Patterned photoresponsive living composite with the experimental yeast strain in media lacking L-histidine where blue light is first targeted on the left side and then the right side (Scale bar: 10 mm). Each data point represents the mean ( $n=3$ ), and error bars represent standard deviation.