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# Preadsorbed SO<sub>3</sub> Inhibits Oxygen Atom Activity for Mercury Adsorption on Cu/Mn Doped CeO<sub>2</sub>(110) Surface

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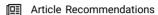


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ABSTRACT: The coadsorption of  $Hg^0$  and  $SO_3$  on pure and Cu/Mn doped  $CeO_2(110)$  surfaces were investigated using the Density Functional Theory (DFT) method. A p (2 × 2) supercell periodic slab model with seven atomic layers was constructed to represent the  $CeO_2(110)$  surface. The results indicated that  $Hg^0$  physically adsorbed on the  $CeO_2(110)$  surface, while  $Hg^0$  chemically adsorbed on the Cu/Mn doped  $CeO_2(110)$  surface, which agree well with the experimental results that Cu and Mn doped  $CeO_2$  greatly improved the  $Hg^0$  adsorption capacity of the adsorbent. The calculated results suggested that  $SO_3$  more easily adsorbs on the above three surfaces than  $Hg^0$  due to the higher adsorption energy. The adsorption configurations and electronic structures indicated  $SO_3$  reacted with O atoms of the surface to form  $SO_4^{2-}$  species. Hence,  $SO_3$  inhibits  $Hg^0$  adsorption on the  $CeO_2(110)$  surface by competing with  $Hg^0$  for surface lattice oxygen. In addition,  $SO_3$  decreased the activity of the surface O atoms, which directly caused the negative effect on  $Hg^0$  adsorption.

## 1. INTRODUCTION

Mercury is a heavy metal element with volatility, migration, and bioaccumulation, which has received a great deal of attention in recent years. In coal-fired flue gas, mercury has three basic states, elemental mercury (Hg<sup>0</sup>), oxidized mercury (Hg<sup>2+</sup>), and particulate mercury (Hg<sup>p</sup>).<sup>2,3</sup> It is difficult to remove Hg<sup>0</sup> using existing air pollution control devices since the elemental mercury has low solubility and reactivity. Selective catalytic reduction (SCR) catalysts can oxidize Hg<sup>0</sup> to Hg2+, and then Hg2+ is captured by wet flue gas desulfurization (WFGD) due to its solubility. 4,5 Therefore, the combined utilization of SCR catalysts and the WFGD system is an effective technology for Hg<sup>0</sup> removal. Metal oxides such as V<sub>5</sub>O<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>, and Al<sub>2</sub>O<sub>3</sub> have been extensively studied due to their high levels of catalytic activity. 6-8 CeO<sub>2</sub> is also an excellent candidate with sulfur resistance, nontoxicity, and low-cost. 9-11 The interchange between Ce3+ and Ce4+ in CeO<sub>2</sub> plays an important role in its redox properties, 12-14 while the pure CeO<sub>2</sub> exhibited poor catalytic performance due to limited oxygen storage capacity. Previous research has indicated that doping with heteroatoms on CeO2 can greatly improve its catalytic activity. 15-17 Copper (Cu) and manganese (Mn), as transition metals, react with the CeO<sub>2</sub> matrix to generate Cu-O-Ce or Mn-O-Ce structure, which exhibit high catalytic reactivity.<sup>8,18</sup> Wang et al.<sup>19</sup> found that Mn doping significantly enhanced the surface activity since surface oxygen vacancies were generated on the surface of Mn doped CeO<sub>2</sub>(111). Guo et al.<sup>20</sup> found that the energy barrier of CO<sub>2</sub> dissociation on the Cu doped CeO<sub>2</sub>(111) surface was only 0.90 eV, which was much lower than that on the perfect CeO<sub>2</sub>(111) surface, 3.70 eV. CeO<sub>2</sub> has three stable low index surfaces:  $CeO_2(110)$ ,  $CeO_2(111)$ , and  $CeO_2(100)$ . <sup>21,22</sup>

Previous studies have shown the performance of  $\mathrm{Hg^0}$  adsorbed on metal doped  $\mathrm{CeO_2(111)}$  surfaces. Compared with  $\mathrm{CeO_2(111)}$ , the  $\mathrm{CeO_2(110)}$  surface presents some unique properties. The  $\mathrm{CeO_2(110)}$  surface has a higher surface energy and lower oxygen vacancy formation energy than the  $\mathrm{CeO_2(111)}$  surface. In addition, the  $\mathrm{CeO_2(110)}$  surface has a unique open plane structure which could provide different adsorption sites for mercury.

The SO<sub>3</sub> concentration is much higher than Hg<sup>0</sup> in the flue gas, hence the presence of SO<sub>3</sub> will have a certain effect on the removal of Hg<sup>0</sup>. Many previous experimental studies have shown that SO<sub>3</sub> affects the catalytic oxidation of Hg<sup>0</sup>. Yang et al.26 found that CeO2 oxidized SO2 to form SO3, which subsequently reacted with Hg0 to produce HgSO4 under simulated flue gas. Zhuang et al.<sup>27</sup> found that SO<sub>3</sub> inhibited the oxidation of mercury; the oxidized mercury in the SCR outlet decreased from 71% to 45% when 50 ppm of SO<sub>3</sub> was added to the flue gas across the SCR. Sjostrom et al.<sup>28</sup> pointed out that mercury capture decreased from 85% to 17% after the addition of 10.7 ppm of SO<sub>3</sub>. However, Cao et al.<sup>29</sup> indicated that the Hg<sup>0</sup> oxidation efficiency increased by approximately 20% when adding 50 ppm of SO<sub>3</sub> to flue gas in the SCR slipstream reactor. From a theoretical aspect, He et al.<sup>30</sup> indicated that SO<sub>3</sub> decreased the adsorption energy of Hg<sup>0</sup> on a carbonaceous surface since SO<sub>3</sub> suppressed the activity of its

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next-nearest-neighbor carbon atom. However, few theoretical studies investigated the adsorption mechanism of  $Hg^0$  and  $SO_3$  on the  $CeO_2(110)$  surface. DFT calculation has been widely used in the study of  $Hg^0$  adsorption on different metal or metal oxide surfaces. This simulation results can also provide additional information with the experimental results. Thus, combined, the experimental and simulation calculation will be the best way to better understand the mechanism of mercury adsorption.

In this study, the DFT method was conducted to investigate the effect of  $SO_3$  on  $Hg^0$  adsorption on pure and Cu/Mn doped  $CeO_2(110)$  surfaces. The adsorption energy, adsorption configuration, and electronic structure were calculated to study the effect of  $SO_3$  and Cu/Mn doping on  $Hg^0$  adsorption. The mercury adsorptions on  $CeO_2$ ,  $Cu/CeO_2$ , and  $Mn/CeO_2$  catalysts were conducted by experimental methods to further study the  $Hg^0$  adsorption performance on different catalysts.

#### 2. COMPUTATIONAL AND EXPERIMENTAL SECTION

**2.1. Computational Method.** In this study, all the density functional theory calculations were performed using the DMol<sup>3</sup> software. The exchange-correlation potential was calculated by the GGA-PBE method. The double numerical basis set, plus polarization with p-functions (DNP), was applied for the molecular orbitals. The core DFT semicore pseudopotential (DSPP) method was used to set the core treatment of Hg, Cu, and Ce, while the allelectron method was applied to O and S. Spin-polarized geometry optimization and a 5.0 Å global orbital cut off was used during the calculations. A 0.005 Ha smearing was applied to increase the computing speed. Three convergence criteria were used for geometric optimization: (1) a maximum force tolerance of 0.002 Ha/Å, (2) a maximum displacement tolerance of 0.005 Å, and (3) a maximum energy tolerance of  $1 \times 10^{-5}$  Ha.

**2.2. Computational Models.** The crystal configuration of  $CeO_2$  is a face-centered cubic structure with a space group  $Fm\overline{3}m$ , as shown in Figure 1A. The Monkhorst-pack grid parameters of the unit cell are  $4 \times 4 \times 4$ . The optimized unit cell parameters (a = b = c = 5.465 Å) were within 1.1% error of the experimental lattice constants (a = b = c = 5.411 Å).<sup>35</sup>

In comprehensive consideration of calculation accuracy and computing resources, a p (2  $\times$  2) supercell periodic slab model with seven atomic layers was constructed, as shown in Figure 1B. The lower two atomic layers are fixed, and the upper five atomic layers are fully relaxed. A 15 Å thick vacuum region was set so that the energy effect of interactions between the slabs can be neglected. The Monkhorst-pack grid parameters of the CeO<sub>2</sub>(110) surface are 3  $\times$  3  $\times$  1. There are two kinds of adsorption sites: the surface sites (X<sub>suf</sub>) and the subsurface sites (X<sub>sub</sub>). The eight adsorption sites, O<sub>suf</sub>, O<sub>sub</sub>, Ce<sub>suf</sub>, Ce<sub>sub</sub>, hollow<sub>sub</sub>, hollow<sub>suf</sub>, Ce-O<sub>sub</sub>, and Ce-O<sub>suf</sub> are shown in Figure 1C. The surface O atoms at the different locations are numbered (O(1) to O(8)) to facilitate following the calculation and analysis, as shown in Figure 1C.

Correcting the f orbital of Ce by Hubbard parameter U can describe the electronic structure of  $CeO_2$  more accurately. However, Kumari et al.  $^{36}$  stated that the effect of Hubbard parameter U correction on the oxygen vacancy formation energy of the stoichiometric  $CeO_2(110)$  surface is slight. Some research shows that using only a DFT method without Hubbard parameter U correction can also provide a reasonable prediction of mercury adsorption on the  $CeO_2$  surfaces.  $^{23,24}$  Therefore, the calculation method of DFT without Hubbard parameter U correction is selected in this study.

 $SO_3$  is calculated by using the same geometric optimization parameters as  $CeO_2$  in a  $10 \times 10 \times 10$  Å cell. The calculated bond lengths of S–O are 1.46 Å and the three bond angles are 119.98°,

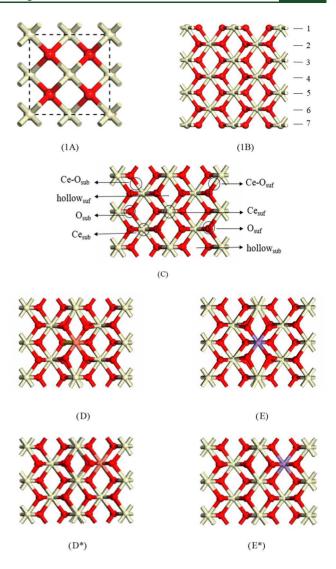


Figure 1. (A) CeO<sub>2</sub> unit cell; (B) side view of CeO<sub>2</sub>(110); (C) top view of CeO<sub>2</sub>(110); (D) top view of Cu-CeO<sub>2</sub>(110); (E) top view of Mn-CeO<sub>2</sub>(110); (D\*) top view of Cu-CeO<sub>2</sub>(110)\*; (E\*) top view of Mn-CeO<sub>2</sub>(110)\*. (The red balls stand for O; the white balls stand for Ce; the pink balls stand for Cu; the purple balls stand for Mn).

 $119.95^\circ$ , and  $120.07^\circ$ , respectively. The calculation parameters of SO<sub>3</sub> are in good agreement with the previous computational values. <sup>37</sup>

There are two kinds of atoms exposed to the environment on the  $CeO_2(110)$  surface, which is shown in Figure 1C as  $Ce_{suf}$  and  $Ce_{sub}$ . To study the effect of metal doping on  $Hg^0$  adsorption, Cu and Mn atoms take the place of the Ce atoms in the first  $(Ce_{suf})$  and second  $(Ce_{sub})$  atomic layers of the  $CeO_2(110)$  surface, respectively. Then, the Cu and Mn doped  $CeO_2(110)$  surfaces were optimized by the same computational parameters as the  $CeO_2(110)$  surface. The optimized Cu and Mn doped  $CeO_2(110)$  surfaces are shown in Figure 1D to E. Figure 1D and E show that the Cu and Mn atoms take the place of the Ce atoms in the first atomic layer of the  $CeO_2(110)$  surface, respectively, denoted as  $Cu-CeO_2(110)$  and Mn- $CeO_2(110)$ . Figure 1D\* and E\* show that the Cu and Mn atoms take the place of the Ce atoms in the second atomic layer of the  $CeO_2(110)$  surface, respectively, denoted as  $Cu-CeO_2(110)$ \* and  $CeO_2(110)$  surface, respectively, denoted as  $Cu-CeO_2(110)$ \* and  $CeO_2(110)$  surface, respectively, denoted as  $Cu-CeO_2(110)$ \* and  $CeO_2(110)$ \*.

**2.3. Computational Parameters.** The adsorption energy  $(E_{\rm ads})$  is defined as follows:

$$E_{\text{ads}} = E_{\text{slab}+X} - E_{\text{slab}} - E_{X} \tag{1}$$

where  $E_{\rm slab+X}$  represents the total energy of the X molecule adsorbed by the substrate model and  $E_{\rm X}$  and  $E_{\rm slab}$  are for the total energy of the X molecule and substrate model, respectively. A negative adsorption energy indicates that the absorption process is an exothermic reaction, while a positive value indicates an endothermic reaction. The adsorption energy was calculated at 0 K by default, and the more negative the adsorption energy, the easier the reaction process will occur.

Hirshfeld population was used to determine the charge distribution of atoms in adsorption configurations. It suggests that the number of electrons around the atom is larger than that of the nuclear charges when the Hirshfeld charges are less than zero; hence, the atom becomes electronegative. Conversely, the atom is electropositive when the Hirshfeld charges are greater than zero.

**2.4. Experimental Method.** In order to verify the simulation results of mercury adsorption on the catalyst surface, CeO<sub>2</sub>, Cu/CeO<sub>2</sub>, and Mn/CeO<sub>2</sub> catalysts were prepared to study the process of mercury adsorption, as shown in Figure 2. The Cu/CeO<sub>2</sub> and Mn/

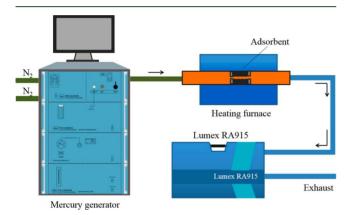


Figure 2. Schematic diagram of Hg<sup>0</sup> adsorption on catalysts.

CeO<sub>2</sub> were prepared using an incipient wetness impregnation method. The CeO<sub>2</sub> powder and Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O mixture or 50% Mn(NO<sub>3</sub>)<sub>2</sub> solution was placed in a beaker and dissolved in deionized water. The resulting solutions were then magnetically stirred for 12 h and further dried at 105 °C for 24 h. This was followed by calcination in air at 550  $^{\circ}\text{C}$  for 3 h. Finally, the obtained catalyst was crushed to a size of 40-60 mesh. The designated Cu/Ce and Mn/Ce molar ratios were 5:100, and the obtained catalysts were abbreviated as Cu/CeO2 and Mn/CeO2. The experimental setup and Hg<sup>0</sup> adsorption efficiency are described in our previous research.<sup>3</sup> Briefly, 0.4 g of the catalyst was fixed with quartz wool in a quartz tube reactor which was placed in an electric resistance furnace, and the experiment was carried out at 200 °C. The gas flow rate of the experiment was 1 L/min, and the initial concentration of Hg<sup>0</sup> was 20  $\mu g/m^3$ . Hg<sup>0</sup> was delivered by N<sub>2</sub> from PSA Cavkit and measured by an online mercury analyzer (RA 915M, Lumex, Russia).

#### 3. RESULTS AND DISCUSSION

**3.1.** Hg<sup>0</sup> Adsorption on the CeO<sub>2</sub>(110) Surface. The  $Hg^0$  adsorption mechanism on the  $CeO_2(110)$  surface was calculated and analyzed. All possible sites that could adsorb  $Hg^0$  were considered. Three stable structures were obtained, which are shown in Figure 3. The corresponding geometry parameters are listed in Table 1. The adsorption energies of the three models are -6.36, -6.95, and -13.82 kJ/mol, respectively, suggesting that the Hg atom has a physical interaction with the  $CeO_2(110)$  surface. According to the adsorption energies, the most stable adsorption configuration is model 3C where the  $Hg^0$  adsorbed on the location of hollow<sub>sub</sub>, with the bond lengths of Hg-O(2) and Hg-O(6) being 3.567 and 3.588 Å, respectively.

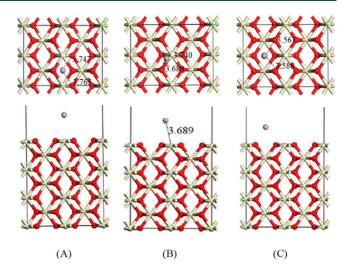
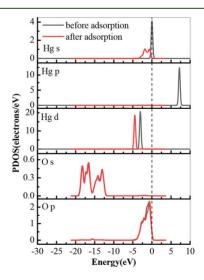


Figure 3.  ${\rm Hg^0}$  adsorption models on  ${\rm CeO_2(110)}$  surface. (The gray balls stand for Hg.)

Table 1. Adsorption Energies, Geometric Parameters, and Hirshfeld Charges for Hg<sup>0</sup> Adsorption on the CeO<sub>2</sub>(110) Surface<sup>a</sup>

configurations	$E_{\rm ads}$ (kJ/mol)	$Q_{\mathrm{Hg}}$ (e)	/			
3A	-6.36	0.101	3.747/3.765			
3B	-6.95	0.121	3.740/3.689			
3C	-13.82	0.122	3.567/3.588			
$^{a}\mathrm{X}$ means atoms on the $\mathrm{CeO_{2}}$ (110) surface.						

The Hirshfeld charges of the Hg atom is within the range of 0.101 to 0.122 eV, demonstrating that few electrons are transferred from  $\mathrm{Hg^0}$  to the  $\mathrm{CeO_2}(110)$  substrate. In addition, the partial density of states (PDOS) can reveal the interaction between different atoms. As depicted in Figure 4, the PDOS of Hg and O(6) atoms in the most stable adsorption configuration (model 3C) are used to explain the interaction between  $\mathrm{Hg^0}$  and the  $\mathrm{CeO_2}(110)$  surface. Compared to the preadsorption, the s, p and d-orbitals intensity of the Hg atom decreased after adsorption due to the transfer of the charges



**Figure 4.** PDOS of Hg and O(6) atoms before and after Hg<sup>0</sup> adsorption in model 3C. The Fermi level ( $E_f$ ) is set to be zero (dashed line in the figure).

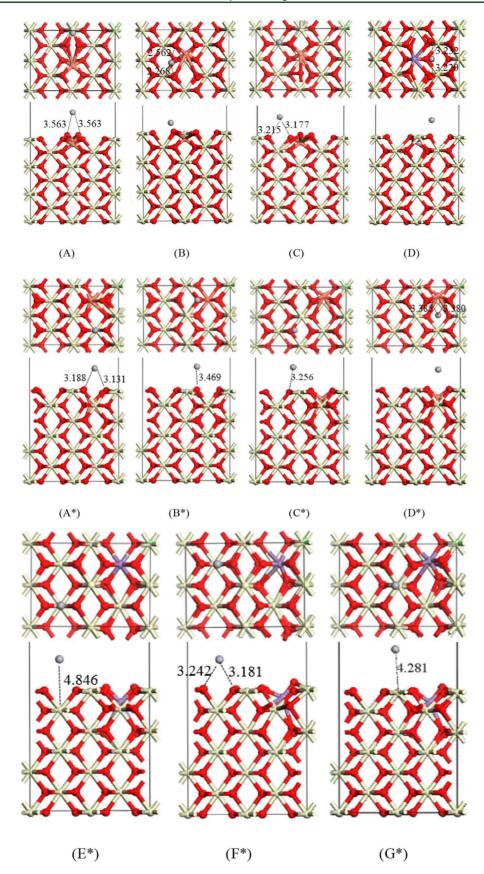


Figure 5.  $Hg^0$  adsorption models on Cu- and Mn-doped  $CeO_2(110)$  surfaces (the figure without \* meaning  $Hg^0$  adsorbed on Cu/Mn- $CeO_2(110)$  surfaces, the figure with \* meaning  $Hg^0$  adsorbed on Cu/Mn- $CeO_2(110)$ \* surfaces).

Table 2. Adsorption Energies and Geometric Parameters for Hg<sup>0</sup> Adsorption on the Cu/Mn-CeO<sub>2</sub>(110) and Cu/Mn-CeO<sub>2</sub>(110)\* Surface<sup>a</sup>

configurations	$E_{\rm ads}$ (kJ/mol)	R(X-Hg) (Å)	configurations	$E_{\rm ads}$ (kJ/mol)	R(X-Hg) (Å)
5A	-182.94	3.563/3.563	5A*	-40.87	3.188/3.131
5B	-186.75	2.562/2.268	5B*	-41.67	3.469
5C	-196.97	3.215/3.177	5C*	-140.23	3.256
5D	-153.98	3.232/3.220	5D*	-146.11	3.385/3.380
			5E*	-56.91	4.846
			5F*	-110.74	3.242/3.181
			5G*	-112.55	4.281

<sup>&</sup>lt;sup>a</sup>X means atoms on the surfaces.

from Hg to the  $CeO_2(110)$  surface, while the s and p-orbitals of the O(6) atom on the  $CeO_2(110)$  surface slightly changed, indicating that the  $CeO_2(110)$  surface still maintained a stable structure after interacting with Hg<sup>0</sup>. Both the Hirshfeld charges and PDOS analysis demonstrated that the interaction between Hg<sup>0</sup> and the  $CeO_2(110)$  surface is weak. The experimental results also found that Hg<sup>0</sup> is physically adsorbed on the surface of the  $CeO_2$  catalyst as discussed in experimental section.

**3.2. Effect of Cu/Mn Doping on Hg**<sup>0</sup> **Adsorption.** 3.2.1. Effect of Cu/Mn Doping on the Mechanism of  $Hg^0$  Adsorption on the  $CeO_2(110)$  Surface. In order to promote the adsorption performance of  $Hg^0$  on the  $CeO_2(110)$  surface,  $Ce_{suf}$  and  $Ce_{sub}$  atoms were replaced by Cu and Mn atoms to study the effect of heteroatoms on  $Hg^0$  removal. The adsorption configurations and adsorption energies are shown in Figure 5 and Table 2. The results show that the adsorption energies of  $Hg^0$  on  $Cu-CeO_2(110)$  and  $Mn-CeO_2(110)$  surfaces are -196.97 kJ/mol and -153.98 kJ/mol, much higher than that that on  $Cu-CeO_2(110)^*$  and  $Mn-CeO_2(110)^*$  surfaces. Therefore, the  $Cu-CeO_2(110)$  and  $Mn-CeO_2(110)$  models are used as the substrate due to their higher adsorption energies.

Three stable structures were obtained after the Hg<sup>0</sup> adsorbing on the different sites of the Cu-CeO<sub>2</sub>(110) surface, as shown in Figure 5A to C. The adsorption energies ranged from -182.94 kJ/mol to -196.97 kJ/mol, which is much higher than that on the pure CeO<sub>2</sub>(110) surface. The most stable adsorption structure is 5C, in which Hg<sup>0</sup> adsorbed on the location of the Ce<sub>sub</sub> site with the adsorption energy of -196.97 kJ/mol. Meanwhile, the Ce-O bonds closed to the adsorption sites and Cu atom change obviously after Hg<sup>0</sup> adsorption. The distances of Cu-O(2) and Cu-O(3) are 3.308 and 2.697 Å, elongated by 66.23% and 32.3%, respectively. The bond lengths of Cu-O(6) and Cu-O(7) are 1.876 and 1.870 Å, shortened by 8.0% and 8.3% relative to the relevant value of 2.039 Å. Hg<sup>0</sup> eventually tends to the position of O<sub>sub</sub> no matter where it is placed on the Mn- $CeO_2(110)$  surface, as shown in Figure 5D. The adsorption energy of model 5D is -153.98 kJ/mol. The distances of Mn-O(3) and Mn-O(7) are 3.164 and 3.167 Å, elongated by 55.8% and 54.9%, respectively. The bond lengths of Mn-O(2) and Mn-O(6) are 1.683 and 1.683 Å, shortened by 17.7% and 17.7% relative to the relevant value of 2.044 Å. These results show that the Cu and Mn doping is in favor of Hg<sup>0</sup> adsorption on the CeO<sub>2</sub>(110) surface, and the order of  $Hg^0$  adsorption capacity on different surfaces is  $CeO_2(110)$  <  $Mn-CeO_2(110) < Cu-CeO_2(110)$ . One of the reasons for this phenomenon is the lattice distortions caused by Cu and Mn doping strengthening the Hg<sup>0</sup> adsorption.

In order to study the effect of Cu and Mn doping on the Hg<sup>0</sup> adsorption, the PDOS, Valence Band Maximum (VBM), Conduction Band Minimum (CBM), and energy gap of  $CeO_2(110)$ ,  $Cu-CeO_2(110)$ , and  $Mn-CeO_2(110)$  models were calculated, which is shown in Figure 6A and Table 3. The Conduction Band (CB) of three surfaces all showed relatively low values. The CB of the CeO<sub>2</sub>(110) model is mainly composed of the Ce 4f orbital, and the Valence Band (VB) of  $CeO_2(110)$  model near the Fermi level is composed of O 2p, Ce 3d, and Ce 4f. After Cu doping, the p and d orbitals divide a new small peak at 0.8 eV, causing the CB of Cu-CeO<sub>2</sub>(110) to shift to a lower energy than that of the  $CeO_2(110)$  model, and the CBM of Cu-CeO<sub>2</sub>(110) decreased from -4.561 to -5.142 eV. A new peak occurs at -0.2 eV in the 3d orbitals of Mn-CeO<sub>2</sub>(110) model, caused the VB near the Fermi level of Mn-CeO<sub>2</sub>(110) shifted to the higher energy level, and the VBM of Mn-CeO<sub>2</sub>(110) also increased from -5.983 to -5.470 eV. Figure 6B is the PDOS of Ce, Cu, and Mn atoms on  $CeO_2(110)$ ,  $Cu-CeO_2(110)$ , and  $Mn-CeO_2(110)$  surfaces, which can be used to explain the influence of Cu and Mn doping on surface electronic structures explicitly. Cu 3d presents obvious CB and VB moves over the Fermi level to a higher energy; Mn 3d orbitals have a strong peak at the Fermi level, resulting in the energy gap of Cu-CeO<sub>2</sub>(110) and Mn-CeO<sub>2</sub>(110) surface decease from 1.422 eV to 0.667 and 0.769 eV. The narrower energy gap could lead to electronic structure change and strengthen the surface reactivity. 39,40 The results from PDOS analysis and the energy gap both agree well with adsorption energy calculation.

3.2.2. Effect of Cu/Mn Doping on Hg<sup>0</sup> Adsorption on the CeO<sub>2</sub>(110) by Experimental Method. The contact time between flue gas containing mercury and adsorbent is less than a few seconds. Thus, the first capture efficiency data in Figure 7A are the most representative of the mechanism between mercury and adsorbent. It is noticed that Hg<sup>0</sup> adsorption efficiency at the first 5 min for CeO<sub>2</sub>, Mn/CeO<sub>2</sub>, and Cu/CeO<sub>2</sub> is 3%, 32%, and 42%, respectively. It is also demonstrated in the bottom right corner of Figure 7A that the Cu and Mn doped CeO<sub>2</sub> have much higher mercury adsorption capacity than CeO2 adsorbent, and Cu doped CeO<sub>2</sub> has the highest mercury adsorption capacity among the three adsorbents. These results are consistent with the calculated results for the adsorption energy of Hg<sup>0</sup> on the  $CeO_2(110)$ , Mn- $CeO_2(110)$ , and  $Cu-CeO_2(110)$  of -13.82, -153.98, and 196.97 kJ/mol, respectively.

The calculated results also were proved through desorption results in Figure 7B. There was only one mercury desorption peak for  $CeO_2$  at 194 °C, which was corresponding to physically adsorbed mercury. While three desorption peaks appeared at  $Cu/CeO_2$ , the peak at low temperature was

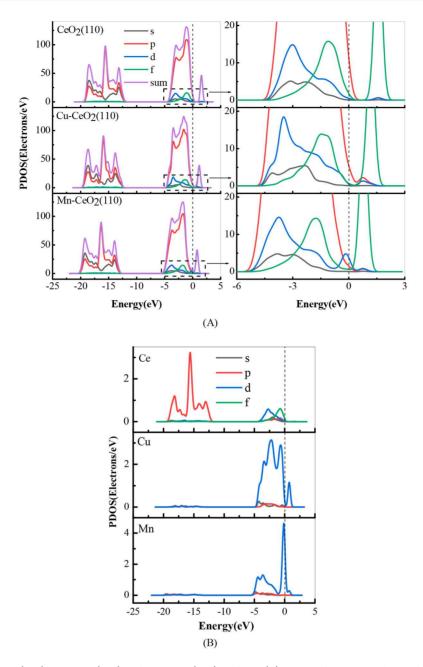


Figure 6. (A) PDOS of  $CeO_2(110)$ ,  $Cu-CeO_2(110)$  and  $Mn-CeO_2(110)$  surfaces. (B) PDOS of Ce, Cu, and Mn of  $CeO_2(110)$ , Cu-CeO<sub>2</sub>(110), and Mn-CeO<sub>2</sub>(110) surfaces. The Fermi level ( $E_f$ ) is set to be zero (dashed line in the figure).

related to physically adsorbed mercury. The desorption peaks located at 246 and 264 °C were attributed to HgO.  $^{38}$  The desorption profile of HgO also presented two peaks at 258 and 281 °C on the Mn/CeO<sub>2</sub> catalyst. The results indicated that the doping of Cu and Mn significantly enhanced the Hg<sup>0</sup> adsorption performance; Hg<sup>0</sup> reacted with the oxygen active site to form HgO on the surface of Cu/CeO<sub>2</sub> and Mn/CeO<sub>2</sub>, which is consistent with the simulated result.

3.3.  $\mathrm{Hg^0}$  and  $\mathrm{SO_3}$  Co-Adsorption on the  $\mathrm{CeO_2}(110)$ ,  $\mathrm{Cu\text{-}CeO_2}(110)$ , and  $\mathrm{Mn\text{-}CeO_2}(110)$  Surfaces. 3.3.1.  $\mathrm{SO_3}$  Adsorption on the  $\mathrm{CeO_2}(110)$ ,  $\mathrm{Cu\text{-}CeO_2}(110)$ , and  $\mathrm{Mn\text{-}CeO_2}(110)$  Surface. The  $\mathrm{SO_3}$  was attached on all possible adsorption sites of the  $\mathrm{CeO_2}(110)$ ,  $\mathrm{Cu\text{-}CeO_2}(110)$ , and  $\mathrm{Mn\text{-}CeO_2}(110)$  surfaces in the form of verticality and parallelism. After geometric optimization, the stable configurations were

Table 3. Energy Gap of Pure and Cu/Mn Doped CeO<sub>2</sub>(110) Surfaces

	VB edge (eV)	CB edge (eV)	energy gap (eV)
1C	-5.983	-4.561	1.422
1D	-5.809	-5.142	0.667
1E	-5.470	-4.701	0.769

obtained, as presented in Figure 8A to E, and the related adsorption energies and geometric parameters are shown in Table 4.  $SO_3$  molecules prefer to bind with the  $O_{top}$  site with a parallel configuration on the three surfaces, which is shown in Figure 8A, D, and E. Model 8A shows that  $SO_3$  adsorbed on the  $CeO_2(110)$  surface with a adsorption energy of -299.49 kJ/mol, and the S-O bond lengths of  $SO_3$  are 1.460, 1.519, and 1.519 Å, respectively. The distance between the S and

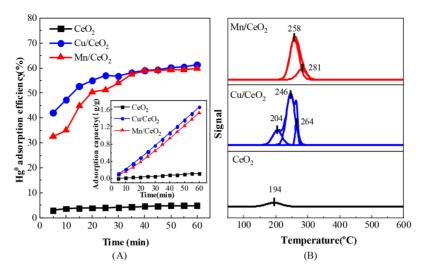


Figure 7. (A) Hg<sup>0</sup> adsorption efficiency of catalysts. (B) Hg<sup>0</sup>-TPD patterns of catalysts.

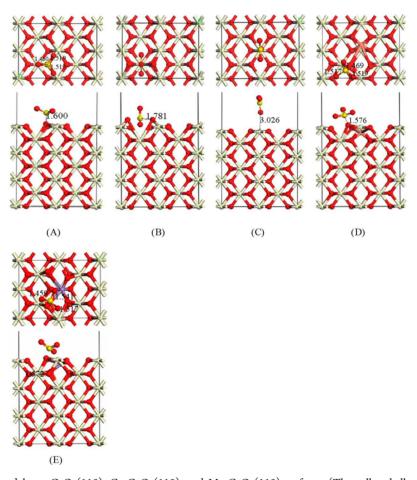


Figure 8.  $SO_3$  adsorption models on  $CeO_2(110)$ ,  $Cu-CeO_2(110)$ , and  $Mn-CeO_2(110)$  surfaces. (The yellow balls stand for S.)

O(6) of  $CeO_2(110)$  surface is 1.600 Å, which approaches the calculated bond length (1.46 Å). This phenomenon indicated that adsorbed  $SO_3$  may have reacted with surface O to generate  $SO_4^{2-}$  species. The adsorption energies of  $SO_3$  adsorbed on  $Cu-CeO_2(110)$  and  $Mn-CeO_2(110)$  surfaces are -427.97 and -445.43 kJ/mol, respectively, which is much higher than that on the  $CeO_2(110)$  surface, demonstrating that  $SO_3$  is inclined to be adsorbed on Cu and Mn doping surfaces. Meanwhile, the distance between the S and O(6) of

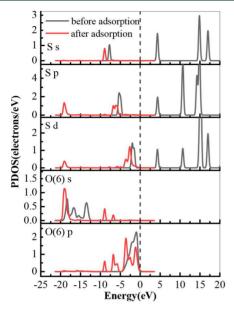
Cu-CeO<sub>2</sub>(110) and Mn-CeO<sub>2</sub>(110) surfaces is 1.576 and 1.572 Å, which is closer than that of the CeO<sub>2</sub>(110) surface (1.600), further proving the above results. The adsorption energies of SO<sub>3</sub> adsorbed on CeO<sub>2</sub>(110), Cu-CeO<sub>2</sub>(110), and Mn-CeO<sub>2</sub>(110) surfaces are much higher than that of Hg<sup>0</sup> adsorption on these three surfaces, indicating that SO<sub>3</sub> is more easily adsorbed on the CeO<sub>2</sub>(110) surface than Hg<sup>0</sup>.

The PDOS analysis of the S and O(6) atoms in model 8A is shown in Figure 9. After adsorption, all orbitals of the S

Table 4. Adsorption Energies and Geometric Parameters for All of Models in Figures 8 and 10<sup>a</sup>

8A -299.49 1.600 10A -8.44	2.417
	3.416
8B -43.57 1.781 10B -12.57	3.649/3.542
8C -11.47 3.026 10C -13.10	3.679/3.540
8D -427.97 1.611 10D -44.38	2.169/2.500
8E -445.43 1.606 10E -14.47	3.764/3/547

<sup>a</sup>X means atoms on the surfaces.



**Figure 9.** PDOS of S and O(6) atoms before and after  $SO_3$  adsorption in model 8A. The Fermi level ( $E_f$ ) is set to be zero (dashed line in the figure).

atoms shifted to lower energy states with an obvious decrease of the PDOS value due to a stronger bond caused by greater hybridization. Specifically, a peak at -19.2 eV appeared on the p and d orbitals, which is strongly hybridized with the s orbitals of the O(6) atom. It was also found that the s and p orbitals of the O(6) atom split into several peaks and overlapped with the s, p, and d orbitals of the S atom at -9, -6.5, and -2.7 eV. All of these results confirmed the strong interaction between S and O(6) on the CeO<sub>2</sub>(110) surface, further proving that SO<sub>3</sub> reacted with surface O to generate SO<sub>4</sub><sup>2-</sup> species.

3.3.2. Effect of  $SO_3$  on the  $Hg^0$  Adsorption on  $CeO_2(110)$ ,  $Cu\text{-}CeO_2(110)$ , and  $Mn\text{-}CeO_2(110)$  Surfaces.  $SO_3$  adsorption on the top of O(6) in model 8A is used as a substrate to study the effect of  $SO_3$  on the  $Hg^0$  adsorption on the  $CeO_2(110)$  surface.  $Hg^0$  was placed to on possible adsorption sites on the  $CeO_2(110)$  surface in the presence of  $SO_3$ . Three stable structures were obtained as shown in Figure 10A, B, and C, and the corresponding adsorption energies are given in Table 4. As shown in model 10C,  $Hg^0$  prefers adsorbing on the location of  $O_{sub}$  with the adsorption energy of -13.10 kJ/mol, which is lower than that on the  $CeO_2(110)$  surface without  $SO_3$  (-13.82 kJ/mol for model 3C). Hence,  $SO_3$  presented a slightly negative effect on the  $Hg^0$  adsorption on the  $CeO_2(110)$  surface.

 $SO_3$  adsorbed on the  $Cu\text{-}CeO_2(110)$  surface (model 8D) is used as the substrate, and model 10D is the most stable structure. In model 10D,  $Hg^0$  was adsorbed on the location of

hollow<sub>sub</sub> with the adsorption energy of -44.38 kJ/mol. Regarding the Mn-CeO<sub>2</sub>(110) surface, SO<sub>3</sub> adsorbed on the top of O(6) in model 8E was used as the substrate. The model 10E is the most stable structure of Hg<sup>0</sup> adsorbed on the Mn-CeO<sub>2</sub>(110) surface in the presence of SO<sub>3</sub>, with an adsorption energy of -14.47 kJ/mol, respectively. It can be concluded that the adsorption energies of Hg<sup>0</sup> on the Cu and Mn doped CeO<sub>2</sub>(110) surface with SO<sub>3</sub> were much lower than that without SO<sub>3</sub> (-196.97 kJ/mol for SA and -153.98 kJ/mol for SD). Therefore, SO<sub>3</sub> inhibited the Hg<sup>0</sup> adsorption on the Cu and Mn doped CeO<sub>2</sub>(110) surfaces.

In order to further understand the effect of SO<sub>3</sub> on Hg<sup>0</sup> adsorption, the Hirshfeld charges of the surface O atoms (O(1) to O(8)) are calculated, listed in Table 5. If the Hirshfeld charge of one oxygen atom becomes more negative, its propensity for Hg<sup>0</sup> adsorption is enhanced.<sup>30</sup> In Table 5, the 1C, 1D, and 1E are the  $CeO_2(110)$ ,  $Cu-CeO_2(110)$ , and Mn-CeO<sub>2</sub>(110) surfaces, respectively. The 8A, 8D, and 8E correspond to the most stable configuration of SO<sub>3</sub> adsorption on  $CeO_2(110)$ ,  $Cu-CeO_2(110)$ , and  $Mn-CeO_2(110)$  surfaces, respectively. On the basis of the above calculations, the ultimate position of Hg<sup>0</sup> was closest to the surface O atoms, indicating that Hg<sup>0</sup> adsorption on the three surfaces is the consequence of interaction between Hg<sup>0</sup> and surface O atoms. The charges of the surface O atoms on model 8A decreased from -0.359 to -0.346 (O(1)), -0.334 (O(2)), -0.342(O(3)), -0.349 (O(4)), -0.338 (O(5)), -0.216 (O(6)), -0.345 (O(7)), and -0.354 (O(8)) in the presence of SO<sub>3</sub>. The effect of SO<sub>3</sub> on the Hirshfeld charge transfer of Cu- $CeO_2(110)$  and Mn- $CeO_2(110)$  surfaces is similar to that on the  $CeO_2(110)$  surface, indicating that  $SO_3$  addition led to the electron transfer among CeO<sub>2</sub>(110), Cu-CeO<sub>2</sub>(110), and Mn-CeO<sub>2</sub>(110) surfaces, particularly reducing the electron accepting ability of surface O atoms; the adsorption capacity of O(1)-O(8) on Hg<sup>0</sup> declined as SO<sub>3</sub> preadsorbed on the surface. Therefore, the existence of SO<sub>3</sub> inhibited the activity of surface oxygen anions in the case of CeO<sub>2</sub>.

On the basis of the above analysis, the presence of  $SO_3$  exhibits a negative effect on  $Hg^0$  adsorption mainly due to the following two reasons. First,  $SO_3$  is more easily adsorbed on the  $CeO_2(110)$  surface than  $Hg^0$ .  $SO_3$  could occupy the active sites prior and competes with  $Hg^0$  when  $SO_3$  and  $Hg^0$  coadsorbed on the three surfaces. Second,  $SO_3$  inhibits the activity of surface oxygen anions in the case of  $CeO_2$ , resulting in the negative effect on  $Hg^0$  adsorption. During the experiment,  $SO_3$  was generated by oxidizing  $SO_2$  over a vanadium—titanium catalyst in the presence of  $O_2$ ; thus, it is difficult to establish an oxygen-free atmosphere to study the effect of  $SO_3$  on  $Hg^0$  without  $O_2$ . The effect of  $SO_3$  on  $Hg^0$  in the presence of  $O_2$  will be studied in further theoretical and experimental research.

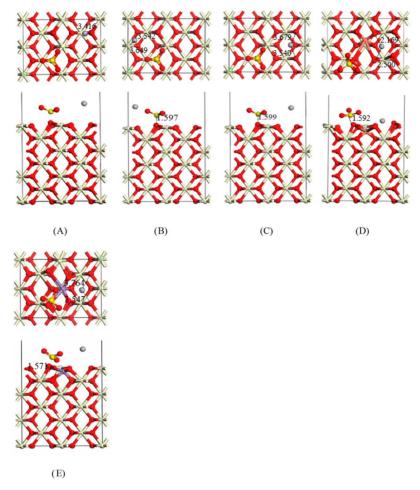


Figure 10. Hg<sup>0</sup> adsorption models on CeO<sub>2</sub>(110), Cu-CeO<sub>2</sub>(110), and Mn-CeO<sub>2</sub>(110) surfaces in the presence of SO<sub>3</sub>.

Table 5. Hirshfeld Charges of Surface O Atoms in Different Models

$Q_{H}(e)$	1C	8A	1D	8D	1E	8E
O(1)	-0.359	-0.346	-0.360	-0.323	-0.361	-0.342
O(2)	-0.359	-0.334	-0.294	-0.291	-0.302	-0.236
O(3)	-0.359	-0.342	-0.294	-0.307	-0.302	-0.282
O(4)	-0.359	-0.349	-0.360	-0.340	-0.361	-0.349
O(5)	-0.359	-0.338	-0.360	-0.318	-0.361	-0.338
O(6)	-0.359	-0.216	-0.294	-0.220	-0.302	-0.175
O(7)	-0.359	-0.345	-0.294	-0.329	-0.302	-0.328
O(8)	-0.359	-0.354	-0.360	-0.332	-0.361	-0.338

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### 4. CONCLUSIONS

The adsorption process of  $SO_3$  and  $Hg^0$  over pure and Cu/Mn doped  $CeO_2(110)$  surfaces was studied by theoretical methods. The calculated results also indicated that the introduction of Cu and Mn strengthened surface relaxation and decreased the energy gap of the  $CeO_2(110)$  surface, resulting in improvement of the catalytic activity of the  $CeO_2(110)$  surface for  $Hg^0$ . The experimental results indicate that the  $Hg^0$  adsorption efficiency is in the order of  $CeO_2 < Mn/CeO_2 < Cu/CeO_2$ , which agree well with calculated adsorption energy. There is only the physical desorption peak for  $CeO_2$ , while the chemical desorption peak is the main form of mercury adsorption on the  $Cu/CeO_2$  and  $Mn/CeO_2$  catalysts, further demonstrating the chemical adsorption of surface oxygen atoms on  $Hg^0$  in the  $Cu-CeO_2(110)$  and  $Mn-CeO_2(110)$  surfaces. The simulation results indicate that  $SO_3$ 

presents a negative effect on  $Hg^0$  adsorption over pure and Cu/Mn doped  $CeO_2(110)$  surfaces. The adsorption energy of  $SO_3$  is much higher than  $Hg^0$ , and the PDOSs of S and surface O atoms hybridize strongly, causing  $SO_3$  to strongly interact with surface active oxygen to form  $SO_4^{\ 2^-}$ . Hence,  $SO_3$  competes with  $Hg^0$  for surface active oxygen sites. Meanwhile,  $SO_3$  preadsorbed on the  $CeO_2(110)$  surface decreases the activity of surface O atoms (O(1) to O(8)), directly causing a negative effect on  $Hg^0$  adsorption.

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#### Notes

The authors declare no competing financial interest.

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