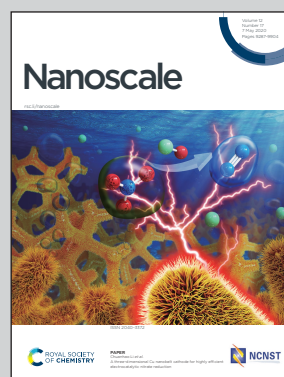


Showcasing research from Professor Choi's laboratory, Department of Chemical and Biomolecular Engineering, Sogang University, South Korea and Professor Lee's laboratory, Department of Chemistry and Chemical Biology, Rutgers University, USA.

Functional nanoarrays for investigating stem cell fate and function

Nanomaterial-based nanoarrays possess the ability to control and sensitively characterize the differentiation of stem cells. This featured review highlights the recent progress made in the synthesis of nanoarrays and provides an overview of electrochemical monitoring methods for controlling and characterizing stem cell differentiation in real-time. The topics discussed in this review aim to give brief and concise guidelines in the development of novel nanoarrays and stimulate interest in live cell study applications.

As featured in:



See Ki-Bum Lee, Jeong-Woo Choi
et al., *Nanoscale*, 2020, 12, 9306.

REVIEW

[View Article Online](#)
[View Journal](#) | [View Issue](#)
Cite this: *Nanoscale*, 2020, **12**, 9306

Functional nanoarrays for investigating stem cell fate and function

 Jin-Ho Lee, ^{†a,b,c} Jeffrey Luo, ^{†a} Hye Kyu Choi, ^b
 Sy-Tsong Dean Chueng, ^a Ki-Bum Lee ^{*a} and Jeong-Woo Choi ^{*b}

Stem cells show excellent potential in the field of tissue engineering and regenerative medicine based on their excellent capability to not only self-renew but also differentiate into a specialized cell type of interest. However, the lack of a non-destructive monitoring system makes it challenging to identify and characterize differentiated cells before their transplantation without compromising cell viability. Thus, the development of a non-destructive monitoring method for analyzing cell function is highly desired and can significantly benefit stem cell-based therapies. Recently, nanomaterial-based scaffolds (e.g., nanoarrays) have made possible considerable advances in controlling the differentiation of stem cells and characterization of the differentiation status sensitively in real time. This review provides a selective overview of the recent progress in the synthesis methods of nanoarrays and their applications in controlling stem cell fate and monitoring live cell functions electrochemically. We believe that the topics discussed in this review can provide brief and concise guidelines for the development of novel nanoarrays and promote the interest in live cell study applications. A method which can not only control but also monitor stem cell fate and function will be a promising technology that can accelerate stem cell therapies.

 Received 30th December 2019,
 Accepted 7th February 2020

DOI: 10.1039/c9nr10963c

rsc.li/nanoscale

^aDepartment of Chemistry and Chemical Biology, Rutgers University, Piscataway, NJ 08854, USA. E-mail: kblee@chem.rutgers.edu

^bDepartment of Chemical and Biomolecular Engineering, Sogang University, Seoul, 121-742, Korea. E-mail: jwchoi@sogang.ac.kr

^cSchool of Biomedical Convergence Engineering, Pusan National University, Yangsan 50612, Korea

[†]These authors contributed equally.

Stem cell therapy is emerging as the next major development in regenerative medicine for functional recovery of tissues and organs damaged by aging, disease, or injuries.^{1–3} Accordingly, an improved understanding of stem cells and control of stem cell fate is likely to benefit treatments for devastating diseases and injuries.^{3–5} However, a significant drawback in the current stem cell therapy is the limited control of stem cell fate, which



Jin-Ho Lee

Dr Jin-Ho Lee is an assistant professor at Pusan National University in the School of Biomedical Convergence Engineering from 2019. Before joining Pusan National University, Dr Jin-Ho Lee received his Ph.D. in Chemical and Biomolecular Engineering from Sogang University (South Korea in 2012) and held postdoctoral appointments at Rutgers University (USA). His research interests span from the synthesis

and integration of nanomaterials for developing bio-inspired platforms for biomedical applications, including stem cell differentiation modulation and biosensor development.



Jeffrey Luo

Jeffrey Luo received his B.S. in Biomedical Engineering from Rutgers University (2016) with minors in Chemistry and Animal Science. As an undergraduate, he developed a nanoparticle-based light-triggered drug release system. He is currently working towards his Ph.D. at Rutgers University in the Chemistry and Chemical Biology department under Dr KiBum Lee's guidance. His research interests include 4D printed devices for high-through-

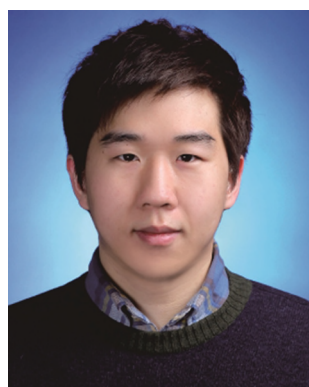
put 3D cell culture and nanomaterial-biomedical scaffold hybrids for enhanced tissue regeneration and therapies.

leads to low efficiency in generating maturely differentiated cells that can replace the damaged original tissues and organs.^{6–8} Addressing the current challenges, researchers have invested in the development of biomaterials that mimic the chemical and mechanical properties of native environments in order to manipulate and control stem cell fate and function in a more precise and defined manner.^{9–12} However, despite the numerous breakthroughs in our understanding of the biological cues that drive stem cell behavior, mimicking native environments remains difficult.

Typically, during stem cell differentiation, cells interact with the surrounding extracellular matrix (ECM), and a variety of intracellular events are influenced by its definable chemical and mechanical properties.^{9,10} Therefore, the mechano-structural properties of the ECM play a significant role in regulating stem cell behaviors. Recently, considerable advancement in nanoarray research has led to the development of sophisticated techniques and materials that can exploit the properties of ECM scaffolds to control cell behavior and repair tissues.^{11,12} While innumerable cell processes differ among different cell types, certain key pathways such as integrin-binding, receptor clustering, and other mechano-transduction cascades remain relatively conserved.¹³ For example, integrin signaling is intimately connected to numerous pathways. Osteogenesis (formation of osteoblasts and bone) has been correlated with stiff substrates, while adipogenesis (formation of adipocytes and fat) has been correlated with soft substrates.¹⁴ Soft substrates (e.g., an adipose microenvironment) promote an increase in active $\beta 1$ integrin, which in turn causes the internalization of BMPR (BMP receptor).^{15,16} This results in decreased BMP/Smad activity, along with decreased osteogenesis. Interestingly, knockdown of $\beta 1$ integrins has a paradoxical effect of decreasing osteogenesis.¹⁷ On the other hand,

activation of $\alpha 2$ due to cell seeding on stiff substrates leads to an increase in osteogenesis.^{16,18} The role of $\alpha 2$ in osteogenesis is further supported by prior reports whereby knockdown of $\alpha 2$ was found to decrease YAP/TAZ translational activity, which decreases osteoblast formation.^{19,20} The at-times contradicting and constantly evolving understanding of substrate stiffness and stem cell differentiation underlines the interplay and complexity associated with mechano-transduction pathways. To this end, mechano-transduction pathways between nanotopography and cell interactions will be thoroughly discussed in this review in order to examine the interaction between substrate topography and cell behavior and enable designing of nanoarrays for pathway-specific stem cell differentiation. A discussion regarding basic biological mechanisms will help lay the foundation for understanding the factors that influence how and why stem cells respond to substrates, including surface chemistry and substrate topography.^{21–25}

One of the current hurdles for most cell characterization methods, such as qPCR and immunostaining, is the requirement of a destructive step, which hinders the safety and effectiveness of stem cell-based therapy.^{26,27} To this end, nondestructive characterization techniques have gained interest for monitoring stem cell behaviors while maintaining a healthy cell population for transplantation.^{28–31} Specifically, functional nanoarrays have gained tremendous attention due to expanding efforts to integrate the concepts of nanochemistry and topography into clinically useful applications.^{28,29} While biological studies continuously elucidate the role of substrate topography in influencing clinically advantageous stem cell behavior (e.g., enhanced differentiation), nanochemistry enables the development of more accurate and sensitive detection methods (e.g., signal enhancement). Therefore, there is significant interest in developing devices that leverage these unique



Hye Kyu Choi

Hye Kyu Choi received his M.S. in Biomedical Engineering from Sogang University (2017) with expertise in bioelectronics. As a Ph.D. candidate at Sogang University in the Chemical and Biomolecular Engineering department (advisor: Prof. Jeong-Woo Choi), his current research interests include the fabrication of a multifunctional microenvironment for stem cell differentiation and biosensing and a muscle tissue-based biorobot for the improvement of conventional drug screening techniques.



Sy-Tsong Dean Chueng

Dr Sy-Tsong Dean Chueng received his Ph.D. in Chemistry and Chemical Biology from Rutgers University (advisor: Prof. KiBum Lee; 2019) with a strong emphasis on the translation of research outcomes into practical applications through interdisciplinary approaches. His research projects have been widely collaborative with the focus being on guiding the differentiation of stem cells through innovative, bio-inspired nanotechnology.

Specifically, he has been involved in the development of various bio-inspired platforms from nanoparticle arrays for neural stem cell differentiation and biosensing, and non-viral nanoparticles for gene expression regulation to the transplantable nanofiber-nanomaterial hybrid scaffold for both neural and mesenchymal stem cell differentiation.

advantages into an integrated nanoarray capable of non-destructively monitoring stem cells as they differentiate, through rationally designed topographical cues.

Although many reviews address stem cell therapy for use in regenerative medicine,^{8,32,33} the tremendous amount of recent activity on nanoarrays for stem cell therapy and a new, live cell-based monitoring approach warrant a thorough review at this time. This review will provide an extensive analysis of the current state-of-the-art developments in the field. More specifically, emphasis will be placed on (i) recent advances in the synthesis methods of nanoarrays, (ii) design and characterization of advanced nanoarrays that provide environmental insoluble cues (nanotopography effect) for stem cell therapy and (iii) their application to non-destructive characterization methods [Fig. 1]. We envision that this article will inspire interest from various disciplines and highlight an important field wherein great strides are being made towards the use of advanced nanoarray systems in the clinical application of stem cell therapy for regenerative medicine.

1. Nanoarray fabrication methods

Functional nanoarrays have considerably advanced the field of stem cell-based therapy by providing both topographical effects on stem cell behaviors and improving the sensitivity of analytical monitoring systems.^{28,29} Accordingly, different synthesis methods have profound effects on the morphology, compositions, and properties of the generated functional nanoarrays. Due to the ubiquity and unparalleled pattern control offered by various lithography techniques, we will focus on the most well-defined techniques, including hydrothermal deposition, chemical/physical vapor deposition, electrical/electrochemical deposition, and template-assisted nanopattern array generation.

1.1 Hydrothermal deposition methods

The conventional hydrothermal deposition method, based on wet chemistry, typically involves hydrolysis and precipitation reactions from a chemical-based aqueous solution to generate nanostructures such as nanoarrays on various shaped substrates.^{34–36} In general, this facile and cost-effective hydrothermal deposition method has been known to be highly efficient in the generation of various shapes and kinds of metal oxide nanostructures.^{37,38} Typically the geometry and distribution of nanoarrays can be varied by altering experimental parameters, including the pH value, temperature, reaction time, and chemical concentration, including those of the precursor. For example, Gao *et al.* successfully synthesized well-oriented double-layered lanthanum-doped SnO₂ nanoarrays by a substrate-free hydrothermal deposition method.³⁹ By adjusting the lanthanum chloride (LaCl₃·6H₂O) concentration, a controlled morphology and phase structure can be obtained. Occasionally, the deposition of seeds as a prior step could facilitate the nucleation and growth of nanoarrays on substrates. To explore this effect, Cook *et al.* hydrothermally synthesized ZnO nanowire on seedless and ZnO-seeded graphene and investigated the effects on the structural geometries.⁴⁰ On the other hand, to scale-up this hydrothermal deposition-based process, Wang *et al.* implemented a continuous flow technique to synthesize uniform ZnO nanorod arrays on 3-D honeycomb substrates.⁴¹ The growth of ZnO nanorod arrays is similar to typical hydrothermal processes. The ZnO seed layer was deposited on wall surfaces of multi-channeled monolithic substrates followed by hydrothermal growth of large-scale ZnO nanorod arrays. However, comparably, significantly improved uniformity and alignment were achieved through integration with a continuous flow technique. The continuous flow-assisted mass transport in a confined space significantly enhanced nucleation and the growth rate of the nanostructure



Ki-Bum Lee

Dr KiBum Lee is a professor at Rutgers University in the Chemistry and Chemical Biology Department. Dr Lee's group has primary interest in developing and integrating nanotechnologies and functional chemical genomics to modulate signaling pathways in stem cells towards specific cell lineages or behaviors. In particular, his group is exploring critical problems in cancer research and stem cell biology to control cell-micro-

environmental interactions at the subcellular and single-cell levels using chemical biology and nanotechnology. From this research effort, he has developed innovative technology platforms that may overcome the critical barriers to harnessing the full therapeutic potential of stem cells.



Jeong-Woo Choi

Jeong-Woo Choi is a full professor at the Department of Chemical & Biomolecular Engineering of Sogang University in South Korea. He received his Ph.D. from Rutgers University (1990), D.Eng. from Tokyo Institute of Technology (2003) and DBA from the University of Durham (2017). His major research areas are nanobioelectronics, brain-on-a chip and bio-hybrid robots.

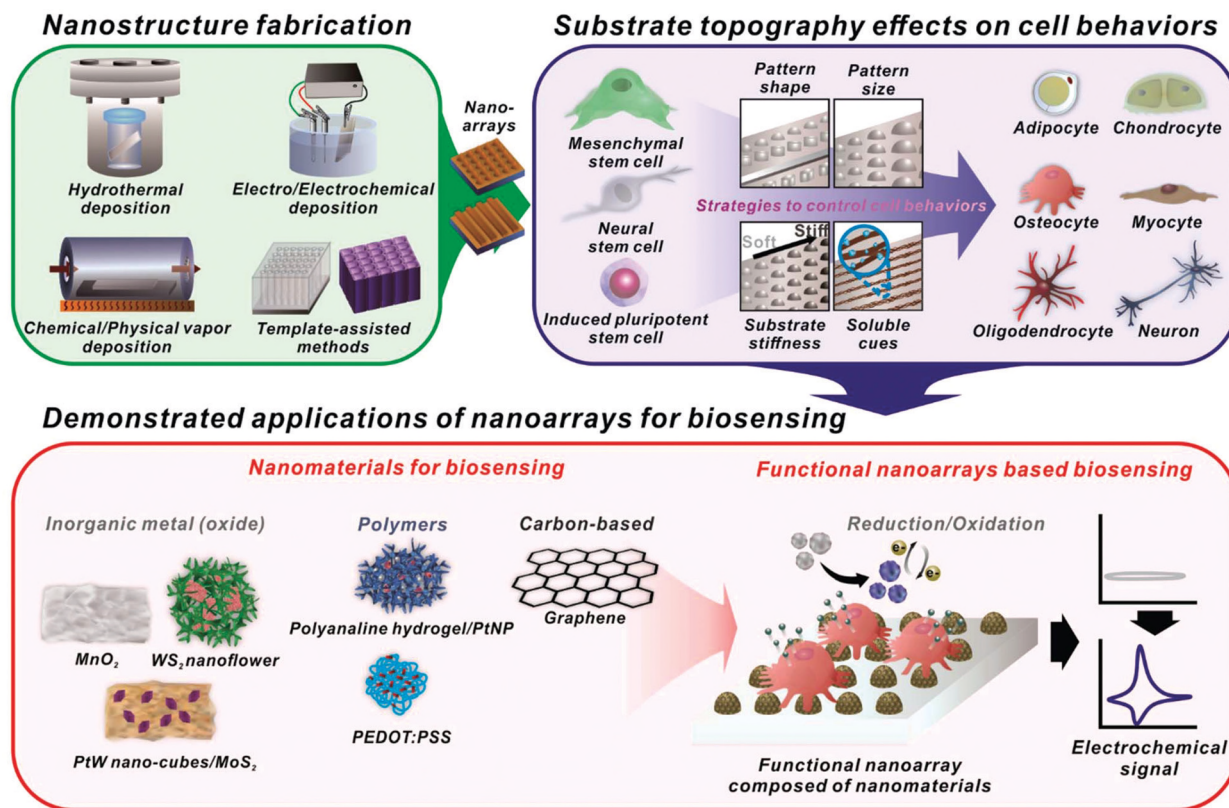


Fig. 1 Schematic illustration of functional nanoarrays for investigating stem cell fate and functions.

on the substrates. Overall, the hydrothermal deposition-based method has proved to be highly efficient, facile, and cost-effective for the growth of various nanostructures on a large scale.

1.2 Chemical/physical vapor deposition methods

Chemical/physical vapor deposition (CVD/PVD) is a vapor phase deposition process wherein the desired material is produced by surface exposure and reaction of gaseous precursors on the surface of the substrate.^{42–44} It has been widely employed to fabricate nanoarrays with high crystallinity and fewer defects. These processes typically require the precise control of various experimental parameters such as pressure, temperature, and precursor concentration. Generally, to obtain high-crystallinity nanomaterials, high pressure, temperature, and a conductive film layer are required as catalysts. For example, high crystallinity of ZnO nanowires can be generated on a Si substrate at high temperatures with a coated layer of a gold film as a catalyst;⁴⁵ however, the use of metal catalysts tends to contaminate the final products, and the catalyst droplets at the growing end of the nanowires can affect the properties of the final nanostructures. Therefore, the direct growth of ZnO nanowires on a common dielectric substrate, such as SiO₂, with the CVD method is of great interest. Xu *et al.* have grown single-crystalline ZnO nanowire arrays directly on a SiO₂ substrate on a large scale using the CVD

approach.⁴⁶ The as-prepared Si substrate was etched to carve out spikes and valleys. After formation, the ZnO nanoclusters act as nucleation sites, which result in the subsequent growth of ZnO nanowires. Uniform size and morphological distribution of ZnO nanowires were achieved with a preferential growth direction along [001]. Similarly, Chen and coworkers have developed a quasi-aligned single-crystalline 3C-SiC nanowire (3C-SiCNW) array with tailored shapes using nitrogen-doping (N-doping) on a flexible carbon fabric. This result has shown the possibility of the utilization of nanoarray fabrication methods to develop flexible electronics that can be integrated into textiles or used in portable electronics.⁴⁷ As can be seen, this gas-phase technique offers the benefit of higher purity deposition on various substrates; however, high vacuum may still be required for non-volatile substrates, and hazardous by-products such as CO, H₂, and HF can be generated in the process, which should be addressed.

1.3 Electro/electrochemical deposition methods

Electro/electrochemical deposition is a widely adopted method for the development of nanoarrays. Electro/electrochemical deposition involves reduction and oxidation (redox) processes, which result in the formation of insoluble precipitates on the surface of the electrode.^{48–51} The final composition and structural morphology can be easily altered through modulation of several experimental parameters such as the solution compo-

sition, potential voltage, current density, deposition time, or electrode substrate. Wang *et al.* demonstrated electrochemical deposition of thorn-like Ni@TiC nanowire arrays and flake-like Co@TiC nanoarray electrodes, which are fabricated by the electrodeposition of Ni and Co catalysts on TiC nanowire arrays.⁵² Instead of applying a constant potential, Gioia *et al.* used a pulsed potential and generated palladium species on multi-walled carbon nanotubes dispersed in a Nafion membrane.⁵³ A pulsed electrodeposition method is a deposition process wherein an applied potential or current density alternates between two or more potential values. Thus, nucleation and crystallization can be controlled by the potential pulse amplitude and duration, which results in excellent mono-sized dispersion. Since these electro/electrochemical methods can only deposit the material of interest onto a conductive surface from a solution containing the corresponding precursor salts, its application is limited to the production of nanostructures on semiconductor and nonconducting substrates.

1.4 Template-assisted methods

The template-assisted method is considered to be an ideal synthesis technique for the design and development of a highly precise pattern array. In general, the template-assisted approach produces a highly ordered nanostructure and allows for control of both size and structure periodicity in a precise manner.^{54–58} Anodized aluminum oxide (AAO) is one of the most widely utilized materials owing to the ease of fabrication, highly ordered pore structure with a controllable pore size, and the amenable ability to be removed with basic solutions. By adjusting the anodization time, potential, and electrolyte solution composition, nanopores with different lengths and diameters can be generated. Subsequently, on the as-synthesized AAO template, a thin layer of an electrically conductive material, such as silver, can be pre-coated on one side of the AAO template to serve as the current collector.³⁰ Then the electroactive materials can electro/electrochemically deposit into the nanopores to form highly ordered nanoarrays.⁵⁹

However, AAO template-based nanoarray development has been highly restricted to smooth and inert surfaces. To address this challenge, Robatjazi *et al.* presented a new strategy for the fabrication of nanostructures with sub-100 nm diameters on both smooth and rough substrates by preparing free-standing through-hole ultrathin alumina membranes (UTAMs).⁶⁰ UTAMs with highly regular arrays of pores were prepared *via* two-step anodization of high-purity aluminum foil. To obtain UTAM containing pores on both sides without breaking of the structure, a thin layer of polystyrene was coated on the UTAMs, and the aluminum base layer was thoroughly detached. After the removal of the polystyrene layer in chloroform, a free-standing through-hole UTAM floating on water was successfully obtained. By using the obtained UTAM as a template, a highly ordered nanoarray was able to be developed on both smooth and rough substrates [Fig. 2].

Due to difficulties in the development of a heterogeneous nanostructure assembly on a large scale, the majority of recent work has focused on developing a single-component nano-

array. In this regard, a new concept for design and fabrication of binary heterogeneous architecture arrays using a binary pore AAO template has been proposed by Wen and coworkers.^{61,62} By employing an over-etching process to partially expose four edges of one set of nano-components in a binary-pore template and selective deposition, a densely packed, heterogeneous nanostructure array was successfully created. The controlled size and shape, as well as inter-pore spacing, could be used to generate various nanostructure arrays with the assistance of binary-pore templates and deposition. In a different manner, Zhao *et al.* created multi-layer nanoarrays sandwiched by anodized aluminum oxide membranes.⁵⁵ Instead of sacrificing the AAO template, it was preserved as a part of the nanostructure to support the second and third layers of multilayer metal arrays.

2. Effects of substrate nanotopography on cellular behaviors

To fully exploit the clinical potential of cell-based therapies, it is advantageous to characterize and utilize all facets of cell behaviors to promote key beneficial processes such as stem cell proliferation and differentiation. Integrins are well-known cell membrane receptors that play a pivotal role in interpreting physical extracellular cues during numerous cell processes. Each of the 24 known final protein assemblies consists of one α subunit (of which there are 18 types), paired with one β subunit (of which there are 8 types) (heterodimeric molecules).²¹ These final pairings, in turn, respond to different stimuli (*e.g.*, binding to ECM proteins *versus* other cells expressing membrane surface adhesion molecules) and have different effects on cell behaviors. For example, a bone regenerating device may be used to encourage osteogenesis (formation of bone and bone-building cells such as osteoblasts). In this situation, it would be beneficial to stimulate integrins containing the $\alpha 2$ subunit (involved in the modulation of osteogenesis) and hinder activity associated with $\beta 3$ integrins (involved in the modulation of myogenesis) to ensure the occurrence of the intended cell behaviors.⁶³

While integrin signals are involved in a multitude of other signaling pathways, one particularly important and interesting downstream target is YAP/TAZ. Activation of YAP/TAZ *via* stiff surfaces for prolonged periods appears to activate a long-term “mechanical memory”, wherein constitutive activation of YAP remains even after cells are placed on a softer surface.⁶⁴ Thus, integrin signaling during prolonged cell culture (typically done on polystyrene flasks) may carry important implications for future use in applications such as regenerative therapies.

2.1 Synergistic strategies for modulating cell behaviors

While appropriate substrate design plays a pivotal role in manipulating cell activity, it is very common to integrate other elements into cell culture scaffolds to ensure greater control over desired behaviors. These elements may range from

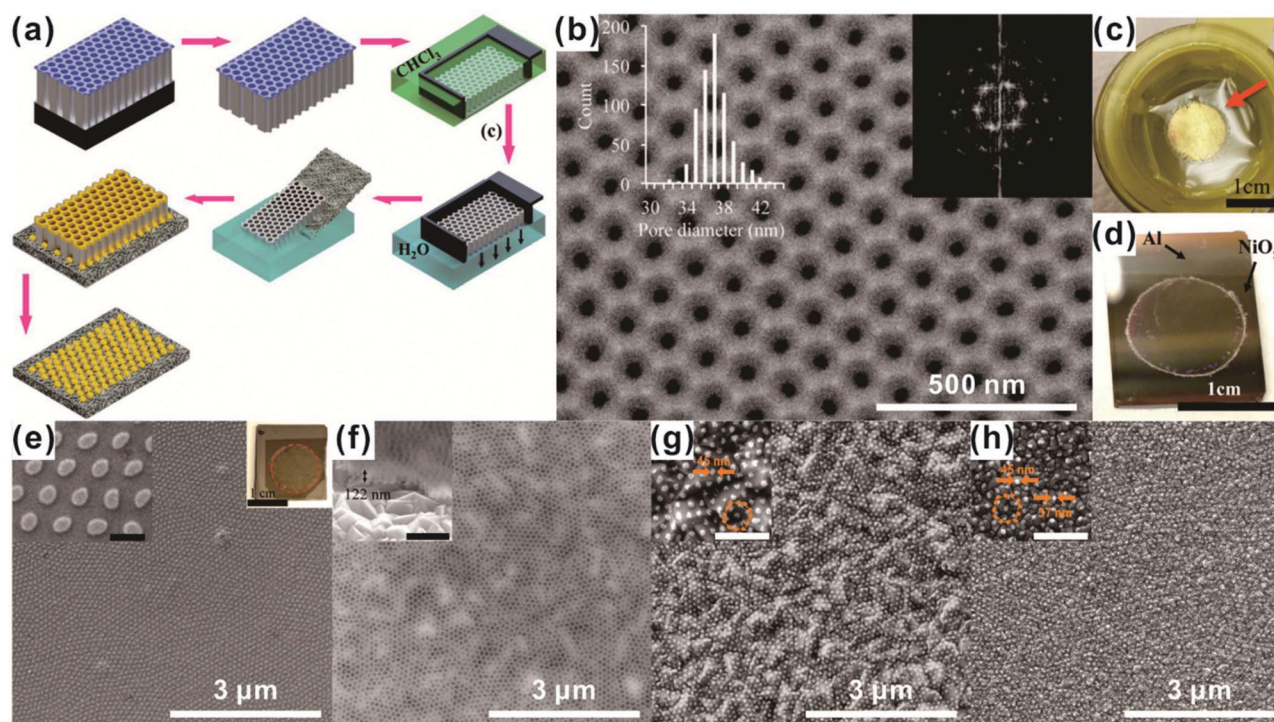


Fig. 2 Fabrication of a nanodot array based on the template-assisted method. (a) Illustration of the fabrication process of an AAO pattern template using ultrathin alumina membranes (UTAM). (b) SEM image of the fabricated template. (c) Digital image of the fabricated template floating on water. (d) Transfer of the fabricated template. (e) SEM image of fabricated nickel nanodot arrays using a template. (f) Plain and oblique views of an SEM image of transferred nanodot arrays. (g and h) Fabrication of gold nanodot arrays using a template-assisted method on an FTO substrate (g) and an ITO substrate (h). Scale bars in the inset images of (e)–(h) are 500 nm. Reprinted with permission from ref. 60. Copyright © 2016 American Chemical Society.

soluble factors in the culture media to external mechanical forces.^{65–70}

Numerous small molecules and growth factors can be incorporated into cell substrates for sustained release during culture. For example, nanofibers can be loaded with stem cell differentiation factors to enable greater control over cell fate. One simple example is the incorporation of TGF- β 1 into chitosan nanoparticles, which were then incorporated into aligned poly-caprolactone (PCL) and poly-L-lactic acid (PLLA) nanofibers.⁶⁷ The nanotopographical cues from the PCL-PLLA nanofibers induced significantly higher expression of smooth muscle markers calponin 1 and SM22 α from seeded MSCs, as compared to a tissue culture plastic control. Notably, the inclusion of TGF- β 1 into the nanofiber, as opposed to supplementation in the cell culture media, enables higher expression of smooth muscle α -actin. Numerous other payloads in other materials can be found elsewhere in literature reports by groups that seek to achieve a similar goal: incorporation of growth factors into their substrates. This concept of incorporating soluble cues (e.g., TGF- β 1 for myogenesis) into substrates with nanotopography (e.g., aligned nanofibers for myogenesis and cell alignment) can enable continued control over cell behaviors after transplantation, since further *in vivo* manipulation can be difficult.

Another widely studied cell control mechanism is the modulation of cell–cell interactions, primarily by changing cell density. For example, chondrocytes and optimal chondrogenesis from MSCs require high densities ($0.5\text{--}1 \times 10^7$ cells per cm^3), leading to efforts to obtain sufficient cells to adequately enable chondrogenesis for large transplantation-scale scaffolds.^{65,66} Li *et al.* alternatively simulated cell–cell interactions by the incorporation of N-cadherin mimetic peptides into fibrillar hydrogels to encourage chondrogenesis.⁶⁸ The authors appended an N-cadherin sequence onto a self-assembling peptide to incorporate the N-cadherin domain onto the fibrillar hydrogel, with minimal effects on the mechanical properties of the substrate. qPCR and western blotting showed significant inhibition of Wnt pathway signaling, leading to increased chondrogenic protein expression.⁶⁸ This effect was short-lived, as most Wnt signaling and chondrogenic protein expression normalized between the N-cadherin-mimicking hydrogels and controls by day 14. Interestingly, the N-cadherin-mimicking hydrogel glycosaminoglycan content (an integral feature of cartilage for maintenance of viscoelastic properties) remained elevated, as compared to controls even after day 14. This approach of simulating cell–cell interactions at lower cell densities can potentially serve to both enhance chondrogenesis and reduce time and materials costs associ-

ated with obtaining the prerequisite number of cells before scaffold seeding.

Another approach that is similar to growth factor incorporation is surface immobilization. Whereas growth factor incorporation may not necessarily lead to a particular spatial arrangement for the payload, surface immobilization implies that the payload is presented on the substrate surface, where it can most readily interact with seeded cells. In one example, myogenesis was enhanced by using 3,4-dihydroxyphenyl-alanine (L-DOPA) to capture sphingosine-1-phosphate (S1P) on nanolithography-patterned polyurethane acrylate (PUA).⁶⁹ The presence of S1P induces greater myogenesis of seeded C2C12 myoblasts. For substrates that undergo harsh processing conditions, which may denature or deactivate sensitive payloads, or are not biodegradable (e.g., encapsulated payloads unable to reach target cells), surface immobilization may be employed to ensure that cells encounter the payload.

Carson *et al.* used a nanoarray with differing line dimensions to examine sarcomere development in cardiomyocytes.⁷⁰ The base substrate, PUA, is non-permissive for induced pluripotent stem cells (iPSC); so RGD domains were chemically grafted to enable cell adherence to the substrate. After a cardiomyocyte differentiation and maturation period, cells cultured on line patterns between 750 and 1000 nm showed significantly longer sarcomeres than cells cultured on a flat surface. While much work is still needed to elaborate on such mechanistic studies, they provide a framework for future efforts that aim to optimize a specific aspect of cell behavior (e.g., nano-fiber substrates which promote cardiomyocyte maturation).

2.1.1 Enhanced, lineage-specific differentiation of mesenchymal stem cells using nanoarrays. Interestingly, substrate stiffness can have significant implications on signaling pathways traditionally associated with soluble cues. For example, chondrogenesis is known to be suppressed when NK- κ B is activated by peptides such as IL-1 β and TNF- α . T. Jiang *et al.* developed PCL-polytetrahydrofuran (PCL-PTHF) nanofibers coated with collagen type 1 to mimic Young's modulus of cartilage.⁷¹ The inclusion of collagen helped suppress the NK- κ B pathway and promote expression of cartilage-specific genes, as compared to nanofibers without collagen. To determine the connection between collagen coating and enhanced chondrogenesis, the authors functionalized atomic force microscope (AFM) cantilevers with integrin subunit β 1, which binds with collagen type 1. Cells cultured on collagen-functionalized nanofibers responded to the β 1-AFM cantilevers similarly to native chondrocytes, while cells on glass or nonfunctionalized nanofibers responded similarly to MSCs. This difference was ablated by treating cells with PF-573228, which inhibits focal adhesions. While further experiments will be needed to confirm the role of integrin β 1 signaling in the NK- κ B pathway, pure topography-based approaches to stem cell differentiation can be advantageous, as they circumvent several problems associated with soluble cues, such as protein degradation and release kinetics (Table 1).

Many load-bearing tissues (*i.e.*, many MSC differentiation lineages) feature specific spatial arrangements and zonal

Table 1 Mesenchymal stem cell differentiation with topography effects of the nanoarray

| Cell | Nanoarray | Differentiation | Ref. |
|------|--------------------------|-------------------------------|------|
| MSC | PCL/PLLA nanofiber | Myogenesis | 67 |
| | Fibrillar hydrogel | Chondrogenesis | 68 |
| | Nanopatterned-PUA | Myogenesis | 69 |
| | PCL-PTHF | Chondrogenesis | 71 |
| | PCL/PLA/PGA nano-grating | Chondrogenesis | 72 |
| | nano-pillar | | |
| | PI nano-patterns | Adipogenesis/ osteogenesis | 63 |
| | HA nanorod | Osteogenesis | 73 |
| | Nanorough substrate | Osteogenesis | 74 |

organizations which assist with *in vivo* function. The design and characterization of substrates that can recapitulate these higher-order architectures from a single cell type can be of clinical value. One group examined nano-grates and nano-pillar topographies on several different materials [PCL, polylactide (PLA), and polyglycolide (PGA) in order of increasing compressive modulus] as a means of generating different zones reminiscent of articular cartilage.⁷² Chondroitin sulfate was included as a surface coating to modulate the polyester surfaces for increased cell adhesion and chondro-inductiveness.⁷² When stress fibers were analyzed by staining, stiffer substrates were found to induce significantly more intense and longer stress fibers than softer substrates.⁷² Additionally, nano-pillar substrates resulted in less intense and shorter stress fibers than nano-grates, potentially due to differences in the curvature.^{72,75} Cells seeded onto stiffer substrates preferentially expressed genes related to deeper cartilage zones: stiff nano-pillars induced expression of ECM proteins common to osteochondral cartilage, while soft and intermediate nano-pillars induced gene expression reminiscent of middle/deep zones. Cells on soft and intermediate nano-gratings expressed proteins found in superficial cartilage.⁷² This substrate stiffness–cartilage zone correlation mirrors natural cartilaginous tissue where deeper cartilage zones (superficial, middle/deep, osteochondral transition) are associated with the greatest resistance to compressive forces.⁷⁶ Unexpectedly, stiff nano-grates (longest stress fiber lengths) seemed to induce noncartilaginous tissue.⁷² Given difficulties in inducing single stem-cell precursors to give rise to multiple controlled cell phenotypes *via* soluble factors (due to shared exposure to the same culture media), rationally designed substrates with graded substrate characteristics are an attractive alternative for the regeneration of tissues with heterogeneous spatial arrangements.

To characterize the MSC response to both micro- and nano-scale topography, G. Abagnale *et al.* utilized two different technologies to generate nano-grating of varying dimensions on polyimide chips.⁶³ Combinatorial micro-scale patterns were generated *via* reactive ion etching, while nano-scale patterns were generated *via* multi-beam interference. On the micro-patterned substrate, MSCs were more apt to differentiate into adipocytes when cultured on wider (15 μ m) ridges, whereas

smaller (2 μm) ridges were most conducive to osteogenesis. Interestingly, staining for vinculin revealed the absence of mature focal adhesions on substrates with smaller micro-ridges. When cultured on substrates monotonically patterned with 450 nm ridges and 200 nm grooves, both adipogenesis and osteogenesis were improved, as compared to a flat control. Despite the known role of YAP/TAZ in adipogenesis/osteogenesis and processes related to mechanotransduction, no difference in YAP cytoplasmic/nuclear localization was observed when comparing nano-grating to flat polyimide substrates. Both micro- and nano-scale patterns enabled cell alignment in parallel with the ridges. This paper provides an interesting distinction where micro-scale topography has vastly different effects (inducing distinct lineage preference) from nano-scale topography (generally increased differentiation efficiency) in a way that complements each other to enhance the final desired cell behavior (*e.g.* efficient generation of functional osteoblasts).

C. Zhao *et al.* tested the combination of micro- and nano-scale patterns on the same substrate for MSC osteogenesis.⁷³ In contrast to the other osteogenesis papers reported here, the authors opted to functionalize the substrate with hydroxyapatite (HA), which is a bioceramic widely used for its osteoinductive and osteoconductive properties.⁷³ Here, the authors developed a technique to compress HA microparticles along a micropattern and allowed HA nanorods to grow under a hydrothermal reaction. Substrates with micro-patterns and nanorods, especially when combined, resulted in higher cell spreading, proliferation, and alkaline phosphatase (ALP) activity than the flat control. Combined micro-patterns and nanorods resulted in elevated or comparable expression for key osteogenic genes BMP2, Runx2, ALP, and collagen type 1, as compared to the topographical features alone; all topographical features resulted in greater expression compared to those of the flat control. The described micro-patterns resulted in elevated $\alpha 5$, αv , and $\beta 1$ expression, while the nano-rods only upregulated αv expression. Examination of specific BMP2 signaling receptors showed that micro-patterns, alone or combined, resulted in increased BMPR1A and BMPR1B, while nano-rods increased BMPR2 expression, indicating potentially different mechanisms at play for different topography dimensions. Unexpectedly, nano-rods alone resulted in the highest osteocalcin expression. This suggests that different topography-length scales may be responsible for activation of different osteogenic pathways, and simultaneous activation of these pathways can result in greater osteogenic efficiency.

Rather than patterning substrates to generate regularly ordered topography, one group of researchers instead used reactive ion etching to create random, heterogeneous “nanorough” (root mean square roughness $R_q = 1$ for flat surfaces to $R_q = 200$ for roughest surfaces) surfaces for MSC osteogenesis [Fig. 3].⁷⁴ Substrates with the highest roughness had the lowest proliferation, even in growth media. Conversely, MSC cultured on the roughest surface exhibited the greatest extent of osteogenesis (measured from ALP, osteopontin, osteocalcin, Runx2, and alizarin red activity) while showing no difference when

directed towards adipogenesis. MSC on flat surfaces displayed larger, sparser focal adhesions that were evenly distributed along the cell area, whereas nanorough surfaces resulted in greater quantities of smaller focal adhesions primarily found on the cell periphery. Actin-myosin/stress fiber organization on nanorough substrates was described as intense and chaotic before aggregation, as compared to initial alignment when cultured on smooth surfaces. This stress fiber arrangement translated to MSC on nanorough surfaces exhibiting greater stiffness than cells on smooth surfaces. One interesting aspect of this research was the effort to determine whether nanorough substrates can replace canonical osteogenic small molecules. Here, the authors investigated dexamethasone, ascorbic acid, and β -glycerophosphate. Osteogenic differentiation media without dexamethasone was found to perform on par with complete differentiation media when cells were cultured on nanorough substrates, indicating that nanoroughness on these substrates can replace the biochemical signaling derived from dexamethasone. Interestingly, qPCR results indicate that ALP, Runx2, and osteopontin expression was upregulated for MSC cultured on nanorough substrates and in the absence of dexamethasone, as compared to complete media. Immunostaining for nuclear/cytoplasmic YAP showed that nanorough substrates were biased for YAP accumulation in the nucleus, which provides a probable mechanism for increased osteogenesis. Treatment with an FAK inhibitor, Y-27632 (reduces stress fibers contractility), and cytoplastin D (reduces actin polymerization) reduced nuclear YAP localization and ALP activity while lysophosphatidic acid, which increases stress fiber formation, had the reverse effect. Altogether, this paper supports the idea that differentiation factors thought to act *via* biochemical pathways may be replaced or supplanted by topographical cues.

2.1.2 Nanoarray-guided mature, functional neural stem cell differentiation. Like MSC, neural stem cells (NSC) respond to substrate stiffness. Adult rat-derived NSC cultured in the same mixed differentiation media (1% FBS, 2 mM glutamate, Neurobasal™ media) will undergo separate differentiation lineages when seeded on substrates with differing stiffness.⁷⁷ Due to the low Young's modulus of native brain tissue, one group used chitosan methacrylamide to generate hydrogels at a physiologically relevant stiffness. Proliferation was maximized on hydrogels with Young's modulus around 3.5 kPa and halted on hydrogels greater than 10 kPa. Additionally, both neuronal and astrocytic differentiation was most prominent when the Young's modulus was less than 1 kPa. For oligodendrogenesis, the authors suggested a biphasic mechanism where a greater number of NSC become oligodendrocytes on stiffer substrates (*e.g.*, 7 kPa) but reach maximum myelination and maturity on soft <1 kPa substrates. The authors suggest that the presence of mature axons and myelination are intertwined, which would severely hinder oligodendrocyte development on stiffer substrates as neuronal development is shown to be most optimal at <1 kPa. Shah *et al.* expanded upon this concept further by generating PCL nanofibers (200–300 nm) to encourage NSC to adopt an oligodendrocyte fate by mimicking

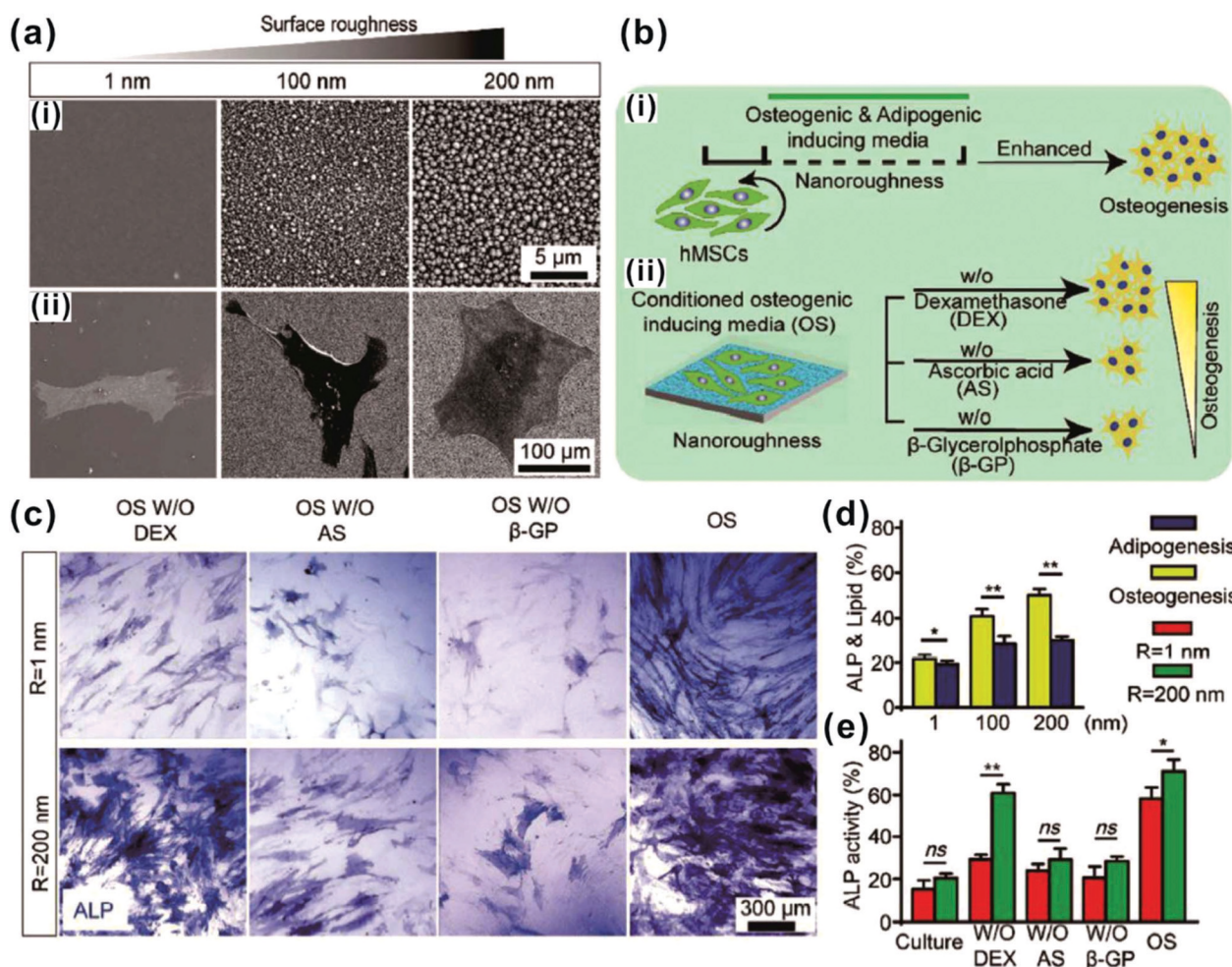


Fig. 3 Enhanced osteogenic differentiation of MSC using a nanotopographic substrate. (a) SEM images of (i) a nanotopographic substrate and (ii) cultured MSC on the substrate. (b) Schematic illustration of the experimental design for osteogenic differentiation of MSC using the nanotopographic substrate. (c) ALP staining of osteogenic differentiated MSC on a smooth and nanorough substrate. (d) Quantification of osteogenic and adipogenic differentiation of MSC on the nanotopographic substrate using ALP staining and oil-red lipid staining. (e) Quantification of osteogenic differentiation for MSC on the nanotopographic substrate in conditioned media. Reprinted with permission from ref. 74. Copyright © 2017 American Chemical Society.

the presence of neuronal axons.⁷⁸ The authors noted that nanofibers, especially when coated with graphene oxide, augmented the expression of key myelination-related genes. These two studies provide an excellent rationale for consideration of substrate stiffness and topography when dealing with neural cell lines (Table 2).

Much of the work done to determine the role of substrates on NSC differentiation has mirrored work on MSC. Factors such as stress fibers, contractility, and pharmacological agents for influencing these processes have been applied to NSC. For example, embryonic stem cells (ESC) induced into neural differentiation exhibit accelerated and aligned stress fiber formation when cultured on nano-grating patterns, as opposed to flat surfaces.⁷⁹ Adding blebbistatin (non-muscle myosin 2 inhibitor) or ML-7 (myosin light chain kinase inhibitor) caused decreases in stress fiber formation and MAP2 (mature neuron marker) expression. A qPCR array for 48 genes related

Table 2 Neural stem cell differentiation with topography effects of the nanoarray

| Cell | Nanoarray | Differentiation | Ref. |
|------|--------------------------|--------------------|------|
| NSC | Hydrogel | Oligodendrogenesis | 77 |
| | PCL nanofiber | Oligodendrogenesis | 78 |
| | Nano-grating | Neurogenesis | 79 |
| | Nano-grating | Neurogenesis | 80 |
| | Nano-grating/nano-pillar | Neurogenesis | 81 |
| | Nanograte/nanopillar | Neurogenesis | 82 |
| | Nanoroughness | Motor-neurogenesis | 83 |

to neurogenesis revealed that 14 genes were upregulated when cells were cultured on nano-grates. Similar to MAP2 staining, the addition of blebbistatin resulted in the expression of these 14 genes returning to basal levels, comparable to a flat control, further supporting the role of stress fiber formation and con-

tractility in neuronal differentiation. Interestingly, Tuj1, an early neuronal differentiation marker, was not affected by the addition of blebbistatin or ML-7.

The roles of nuclear deformation and associated protein lamin A/C were also investigated in the context of flat *versus* nano-grating neuronal differentiation.⁸⁰ ESC cultured with retinoic acid and neuronal differentiation media (positive neurogenesis control) displayed an increase in lamin A/C. Likewise, both MSC and ESC seeded on nano-grating patterns in the absence of retinoic acid for induction of neuronal differentiation showed increases in lamin A/C expression, as compared to the unpatterned controls. The authors also investigated H3K9me1 in MSC as another metric of neurogenesis. Cells seeded on nano-grating substrates showed higher H3K9me1 on days 1 and 4; further culture resulted in no difference between the flat and nano-grating substrates. Notably, this study did not include pharmacological agent addition for establishing whether increased H3K9me1 and lamin A/C levels are directly caused by differing nanotopography, and instead relied on the paper discussed in the previous paragraph to establish the importance of nano-grating in substrate topography-mediated neurogenesis.

One archetypal study seeded induced pluripotent stem cells (iPSC) on nano-grating and nano-pillar arrays with differing widths, spacings, and depths [Fig. 4].⁸¹ A greater nanopattern height (560 nm) resulted in decreased spreading and greater alignment compliance for nano-grating. Concurrently, the greater nanopattern height, regardless of the topographical

shape, resulted in higher cytoplasmic YAP. When combining increased nanopattern height with decreased nanopattern spacing, iPSC exhibited decreased proliferation. In agreement with observations of enhanced alignment and decreased proliferation, iPSC on deep nano-grating substrates had higher expression of neuroectoderm-related (PAX6, day 6) and neuron-related (Tuj1, day 14) genes. Immunostaining data at day 21 showed that both nano-gratings and nano-pillars increase mature neuronal markers (Tau and MAP2). Interestingly, nano-gratings seem to decrease the proportion of glutamatergic neurons and increase GABAergic neurons; however, no mechanistic explanation was offered.

One group generated mutant Rho GTPases to better characterize the enzyme's role in transducing substrate stiffness to NSC.⁸⁴ Dominant-negative RhoA and Cdc42 mutations caused the cells to become less stiff and more inclined to undergo neurogenesis, even on stiffer substrates normally conducive to astrogenesis. Likewise, constitutively active RhoA and Cdc42 caused moderate cell stiffening and bias towards astrogenesis despite culture on softer substrates. The mutant RhoA and Cdc42 cells responded to various contractility-affecting pharmacological agents (*e.g.* Y-27632 and blebbistatin) as anticipated (*i.e.* decreasing contractility rescued neurogenesis on compliant substrates for constitutively active RhoA). Should pharmacological manipulation and substrate stiffness modification be unfeasible, future studies can instead rely on mutant GTPases to enable substrate-independent control over the stem cell differentiation.

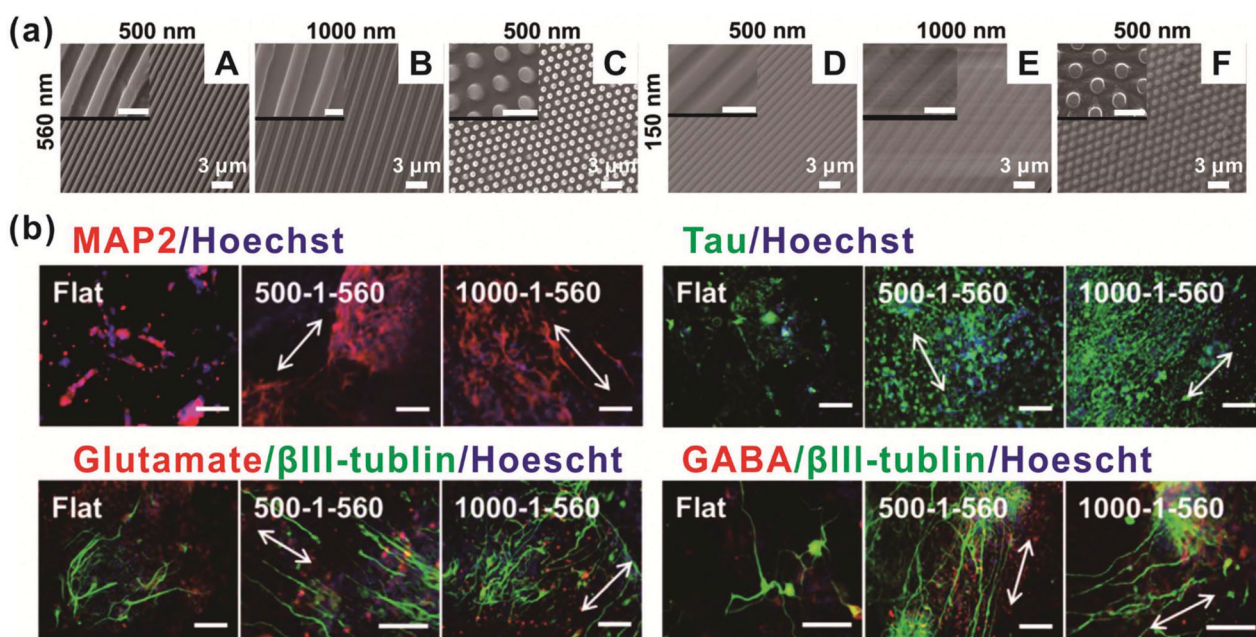


Fig. 4 Neurogenesis of hiPSCs on nanoarrays (nanogratings and nanopillars). (a) SEM images of nanoarrays. Gap size and width size of nanogratings were synchronized as 500 nm (A and D) and 1000 nm (B and E). The diameter of the nanopillar was 500 nm, which was 1.9 times the diameter in center-to-center spacing (C and F). The heights of the nanoarrays were 560 nm (A–C) and 150 nm (D–F). Scale bars in inset images are 1 μm. (b) Expression of neuronal markers after differentiation on nanoarrays. Scale bars are 100 μm. Reprinted with permission from ref. 81. © 2016 Elsevier B.V. All rights reserved.

Another group incorporated a temporal aspect into nanotopography-mediated neurogenesis.⁸² iPSC were chemically induced to dopaminergic neural progenitor cells on nano-grating patterns before transfer onto either another nano-grating or nano-pillar substrate. Tuj1, FOXA2 (a midbrain dopaminergic neuron marker), and tyrosine hydroxylase (an indicator of functional dopaminergic neurons) were found to be upregulated on nano-grating and nano-pillar substrates compared to flat substrates. The authors also sought to determine which nanotopography generated a more mature, “complex” neuron morphology. Nano-gratings resulted in neurons with longer neurite outgrowths, while nano-pillars enabled greater branch terminals, branch points, and dendritic complexity. The electrophysiological analysis showed cells transferred to nano-pillars to have a higher number of cells capable of repetitive spontaneous synaptic activity, as compared to cells on nano-pillar or flat substrates. These results suggest that while nano-gratings are sufficient to prime iPSC for a dopaminergic neurogenic pathway, nano-pillars may enable a more mature neuronal morphology and function.

In a departure from typical nanoarrays with well-defined patterns, one group used reactive-ion etching (RIE) to generate heterogeneous “nanoroughness” substrates on vitronectin-coated glass for hESC and hiPSC.⁸³ Immunostaining after short-term culture (48 hours) showed greatly increased vinculin (associated with mature focal adhesions), actin (stress fibers), and cytoplasmic YAP when cells were seeded onto nanorough surfaces (RMS roughness = 200 nm). Interestingly, cells on nanorough substrates exhibited greatly accelerated conversion from a pluripotency-associated gene expression regimen (e.g., Oct3/4, TERT) to neuroectoderm gene expression (e.g., PAX6, NeuroD1), even when growth (proliferation) media were used. Additionally, cells cultured on nanorough substrates were more responsive to directed differentiation into motor neurons (determined by Olig2⁺, Tuj1⁺, and HB9⁺ staining) compared to cells on smooth controls throughout a 24-day differentiation and maturation period. As a final demonstration, lysophosphatidic acid, which facilitates stress fiber formation, significantly reduces substrate-based differences in YAP localization (i.e., most YAP localized in the nucleus regardless of the roughness), resulting in decreased PAX6⁺ cells. Interestingly, Y-27632 and cytochalasin D both increased cytoplasmic YAP localization (anticipated to increase neurogenesis) but either had no effect or decreased the percent of PAX6⁺ cells (i.e., similar percent of PAX6⁺ cells with or without these pharmaceutical agents for given roughness). An additional understanding of the underlying complexities is needed to fully elucidate the molecular mechanisms regulating neurogenesis.

2.1.3 Nanoarray facilitated maintenance of stem cell pluripotency. Much like any other cells, induced pluripotent stem cells (iPSC) and embryonic stem cells (ESC) are subject to the same mechanotransduction signals and pathways, as discussed previously. Due to considerable clinical potential and instability during *in vitro* culture, the recapitulation of directed differentiation and maintenance of pluripotency are the two

most significant areas of active research.⁸⁵ Since directed differentiation was discussed in earlier sections (MSC and NSC), we will focus on the maintenance of pluripotency as this aspect is unique to this particular class of stem cells.

Traditionally, iPSC have been cultured on mouse embryonic fibroblast (MEF) feeder culture, which helps to condition the cell culture vessel.⁸⁶ With expanding knowledge of basic cell biology, significant efforts have been directed towards transitioning away from xenogenic feeder cell culture to naturally derived ECM-supported culture (e.g., Matrigel® from heterogeneous mouse sarcomas) for clinical translation.⁸⁶ The next goal in this field is to enable iPSC cultured on substrates with wholly human-derived ECM proteins. Kim *et al.* cultured human BJ1 fibroblasts to harvest the ECM deposited on conventional culture vessels.⁸⁶ These resulting ECM-coated vessels were subjected to decellularization and crosslinking with genipin, now termed fibroblast-derived matrices (FDM). Two distinct trends became evident when human ESCs were cultured on FDM with varying degrees of crosslinking and stiffness. Less crosslinked/softer FDM enabled greater cell attachment and proliferation, whereas FDM with higher crosslinking/Young's modulus maintained a greater proportion of alkaline phosphatase-positive (early pluripotency marker) colonies. They concluded that an intermediate-crosslinked FDM created an optimal balance between attachment/proliferation and pluripotency maintenance. This functional iPSC assay is supported by qPCR data, which showed that uncrosslinked FDM decreased the expression of genes coding for Oct4, E-cadherin, and other iPSC-associated proteins and increased the expression of genes coding for N-cadherin, vimentin, and other mesenchyme-associated proteins. In light of these results, the authors suggested that substrate stiffness may affect the ability of iPSC to undergo an epithelial-to-mesenchymal transition. Studies such as these may play a key role in developing commercialized cell therapy manufacturing and compliance plans, where cell line quality assurance is significantly more rigorous than in basic research and development. This development represents an exciting intermediate between *in vitro* MEF feeder culture and the *in vivo* acellular ECM, benefiting from the greater reproducibility inherent to *in vitro* culture, as well as cheaper production and the absence of contaminating feeder cells and human-derived proteins (Table 3).

iPSC and ESC typically grow as colonies on Matrigel or MEF feeder cultures, which are necessary for the maintenance of pluripotency.⁸⁵ Mechanotransduction protein RhoA, which mediates cytoskeletal contractility, is integral to maintaining E-cadherin, a protein involved in cell–cell contact. Loss of E-cadherin can result in apoptosis, disrupted colony for-

Table 3 iPSC with topography effects of the nanoarray

| Cell | Nanoarray | Effect | Ref. |
|------|---------------------|----------------------------|------|
| iPSC | Polyacrylamide gel | Pluripotency | 85 |
| | ECM coated vessel | Proliferation/pluripotency | 86 |
| | PDMS | Pluripotency | 87 |
| | PI/PDMS nano-grates | Colony alignment | 88 |

mation, and loss of pluripotency. Perturbations to RhoA signaling (*e.g.*, TGF- β , which activates RhoA) can cause preferential differentiation (*e.g.*, mesenchymal and endodermal lineages with TGF- β). This is in agreement with the observation that iPSC have lower stiffness and cytoskeletal organization than fibroblasts, chondrocytes, and mesenchymal stem cells. Furthermore, mESC cultured on various polyacrylamide gels (0.6–8 kPa) reveals that softer substrates enable greater expression of the pluripotency gene Oct4 for significantly more cell passages. Caution is advised when designing soft substrates since excessively soft substrates (*e.g.*, 0.1 kPa in one demonstration) may result in decreased viability and proliferation due to an inability to form prerequisite cytoskeletal elements. Additionally, supplementing and withdrawing specific soluble cues may cause cells to display different behaviors in response to the same substrate. In one instance, the removal of FGF-2 and TGF- β from culture media seemed to ablate the iPSC ability to maintain pluripotency based on substrate stiffness.⁸⁵ In another demonstration, honeycomb-shaped nanopattern substrates enabled iPSC to maintain elevated Oct4 expression only in the absence of FGF-2. The addition of FGF-2 caused the opposite effect in this particular report, wherein Oct4 was decreased when cells were seeded on honeycomb nanotopography.

By applying these mechanotransduction principles, Gerardo *et al.* have facilitated the reprogramming of MSC back to iPSC.⁸⁷ Umbilical cord-derived MSC (UC-MSC) were seeded onto tissue culture plastic (GPa range) or polydimethylsiloxane (PDMS) (15 kPa or 1.5 kPa), due to clinical interest owing to the ease of access, relative immaturity/stemness compared to adult stem cells, and less exposure to external mutagens. Cells on stiff substrates presented flatter morphologies, while soft substrates allowed cells to adopt more columnar morphologies with greater cell–cell interactions along the Z-axis. As anticipated, stiffer substrates induced more robust stress fiber formation, larger focal adhesions, higher nuclear mechanical strain, and lower nuclear circularity. When probing the chromatin structure, MSC seeded on softer substrates exhibited lower DAPI fluorescence and higher H4K16ac levels, both of which are correlated with open, euchromatin regions commonly seen in pluripotent cells. To reprogram UC-MSC, polycistronic lentiviruses encoding the four canonical Yamanaka factors were administered to cells on retronectin-coated tissue culture plastic before replating on the aforementioned TCP/PDMS substrates.⁸⁷ MSC on soft substrates generated more iPSC-like colonies (*e.g.*, SSEA-4 and TRA-1-60 positive) with greater reprogramming efficiency than cohorts on stiff substrates. They postulated that increasing euchromatin regions permits greater binding of exogenous reprogramming factors to the target sites, thus accelerating the expression of endogenous pluripotency-maintenance genes. These results indicate that seeding MSC on soft substrates to induce an iPSC-like “relaxed” cytoskeletal and nuclear phenotype prime the cells to undergo accelerated reprogramming.

Since iPSC tend to grow as colonies rather than single cells with limited cell–cell contact, one group sought to characterize the behavior of entire colonies on the nanotopography.⁸⁸ Both

polyimide (periodicity of 650 nm, major study focus) and PDMS (periodicity of 340, 650, and 1400 nm) were employed in this study using light interference lithography to generate nano-grates with high fidelity. Single-cell iPSC aligned along the nano-grates as expected. Interestingly, iPSC within a colony center did not elongate along the direction of the nano-gratings, despite the overall colony displaying alignment. Rather than individual cells aligning, the authors observed cell division planes to be aligned perpendicular to the nano-grating. These findings suggest that nanotopographical surface patterns control the entire colony morphology by influencing mitotic behavior along the colony periphery. PDMS substrates with the periodicity of 650 and 1400 nm mirrored these findings, though cells and colonies on 340 nm substrates notably showed no morphological differences, as compared to flat controls. Additionally, colonies seeded on nano-grating patterns responded more rapidly to BMP4 as a differentiation morphogen (*e.g.*, downregulation of pluripotency genes Oct4 and Nanog) compared to colonies on flat substrates. This differential behavior based on the individual iPSC position within the colony is substantiated by examination of YAP localization. Cells in the interior of the colony showed YAP localization in the cytoplasm, whereas cells on the periphery showed nuclear YAP localization. The significance of this paper is two-fold: (1) documentation of different iPSC behaviors based on the spatial location within a colony, and (2) creating motivation for the examination of other cell types for differential behavior between inner and peripheral cells.

3. Demonstrated applications of nanoarrays for electrochemical biosensing

While many biosensor designs exist and are currently under active investigation and refinement, certain aspects are universal. At a fundamental level, biosensors must be able to (1) recognize and record biological phenomena and (2) transduce that event into an electrical signal for further data processing. Analyte recognition is especially significant with samples that undergo minimal processing (*e.g.*, whole blood samples) and are prone to extraneous biomolecules that mask the signal from the intended analyte. Some of the most popular capture elements include antibodies, oligonucleotides, and enzymes, which are notable for their high sensitivity and selectivity.^{89,90} For prolonged cell culture directly on the sensor, various ECM proteins and cell adhesion moieties may also be considered as capture elements that enable the sensor to interact with cells during culture. Certain applications involving sustained biological phenomena, such as the monitoring of stem cell differentiation over the course of several weeks, or materials that are scarce or costly to obtain, such as primary, freshly extracted cells from non-human primates, may benefit from the ability to read multiple data points from a single sample. Signal transduction is heavily dependent on what detection system is used. The

three major detection systems currently in use are (1) optical, (2) physical, and (3) electrochemical systems.⁸⁹ Optical and electrochemical systems predominate the field owing to non-destructive, live-cell monitoring. Transduction systems may be relatively simple, as in H₂O₂ redox activity measurement on conductive substrates, or complex and involve biological systems, for instance, the cellular detection of biological analytes and secreted electrochemically active molecules.^{29,31,89–95}

Most popular and developed optical systems can be divided into four major subcategories: colorimetric, fluorescent, luminescent, and Raman scattering systems.⁹⁶ Colorimetric transduction systems are the simplest sensing tools, wherein a colored reagent can be observed by the naked eye, leading to rapid detection without complicated instruments. These are most limited by the imprecise nature and poor sensitivity of organoleptic detection. Fluorescence-based systems are the most common optical detection variants. A certain wavelength of light is used to excite a fluorophore to emit photons of a different wavelength, which are then detected. Advantages include high sensitivity, efficiency, and ease of use; disadvantages are primarily related to the specialized equipment, including monochromatic light and filters used to separate excitation light from emitted light. Luminescence-based systems emit light without prerequisite excitation light, resulting in a higher signal-to-noise ratio but still require equipment capable of converting photons into an electrical signal for accurate quantification. Raman scattering-based systems generally rely on surface-enhanced Raman scattering (SERS) to obtain an adequate signal for detection.⁹⁷ Apart from the requirement to tailor substrates specifically for SERS, potentially limiting micro- and nanopography to influence cell behavior, special Raman microscopes must be employed to utilize this technique. Global limitations for optical biosensors include naturally turbid and complex biofluids which may contain optically absorbing and auto-fluorescing biomolecules.⁹⁸

Here, we will focus on electrochemical detection using sensors with some form of micro- or nano-topography. Significant achievements in microelectronic circuit production make these sensors more readily available and translatable for clinical use, eliminating the need for specialized equipment, apart from the sensor itself.⁹⁸ Additionally, electrical signals from these sensors do not require additional transduction equipment for conversion of photons from optical biosensors into electrons *via* charged-coupled device [CCD] cameras to produce a machine-readable signal, which increases performance.⁹⁸ Sensitivities as low as nanomolar concentrations are regularly reported with high selectivity against common biological contaminants, as detailed below. Many biological phenomena of interest, including expression of differentiation markers and key transcription factors, occur at very low concentrations, leading to this review's emphasis on electrochemical detection.

3.1 Roles of nanomaterials in electrochemical biosensing

A variety of electrochemical sensing modalities have been developed, each with advantages and disadvantages.^{21,99}

Controlled design of nanomaterials of different shapes, sizes, arrangements, and compositions can enhance these benefits and compensate for deficiencies, such as a weak signal-to-noise ratio. The exact nature of sensor-nanomaterial/topography interactions varies on a case-by-case basis, though some key concepts are reasonably consistent regardless of the sensor type and target application. Electrochemical detection typically occurs at the sensor interface, which significantly increases with the properties of the nanomaterial (nonparallel surface-area-to-volume) and micro-/nanotopography.^{89,98} The increased surface area in turn typically translates to a higher measurement sensitivity due to (1) a greater proportion of sensor mass affected by the target biological phenomena and (2) greater capture efficiency as a result of more enzymes being present on a greater surface area that can capture a greater number of analyte molecules.^{89,98} Nanomaterials may impart additional functionalities to biosensors, such as enabling cell attachment, enhanced stability against fouling biological molecules, providing additional functional groups for chemical modifications, and improving electrical conductivity.⁹⁸ In many cases, metal-based nanomaterials themselves may directly participate in chemical reactions as catalytic sites.^{89,98} Micro/nanotopographical features can be designed to enable interesting sensor architectures such as 3D porous hydrogel sensors with a dramatically increased surface area, as compared to flat 2D substrates, and enhanced spatial resolution when monitoring different areas of a sensor.^{100,101} Intelligent incorporation of nanomaterials and micro/nanotopography can significantly increase the utility of biosensors by improving performance and permitting additional functionalities. To this end, commonly used biocompatible nanomaterials will be categorized herein, and their uses in biosensors will be described. Key topics for this section will include unique physicochemical features, associated characterization techniques, and limitations for biological applications.^{89,90,102}

3.1.1 Inorganic metals (oxides). Metal nanomaterials have long been used in electrochemical sensing systems, owing to their excellent electrical conductivity, ease of synthesis, and surface modification potential with good biocompatibility.^{103,104} Decoration with metal nanomaterials can enhance the reactivity of the electrode by increasing the active surface area.¹⁰⁵ For example, Hsu *et al.* developed hemispherical pattern arrays on a silicon wafer by photolithography and sputtered Au nanoparticles.¹⁰⁶ The active sensing area was increased 10.2 times compared to that of the planar Au electrode, resulting in an improved detection limit of glucose (9 mM) and high sensitivity (749.2 $\mu\text{A cm}^{-2} \text{ mM}^{-1}$). The oxidative current was stably maintained even after 20 potential cycles. Additionally, metals such as Au, Pt, and Ni and metal oxides such as Fe₃O₄, TiO₂, and NiO are also widely used in electrochemical biosensors as catalysts, which enable nonenzymatic detection of metabolites such as glucose and reactive oxygen species.^{107–111} In particular, owing to the variety of the composites (*e.g.*, Ni, NiO, Ni(OH)₂, *etc.*), nickel was the most intensively examined transition metal for use as a catalyst.¹⁰⁷ In the case of glucose sensing, the redox couple of Ni²⁺/Ni³⁺ is considered to be the reaction center of

glucose oxidation, based on the coincidence of the anodic peak between the transition metal and glucose oxidation. Catalytic surfaces with $\text{Ni}(\text{OH})_2$ (Ni^{2+}) are formed by the glucose oxidation process, followed by the oxidation of $\text{Ni}(\text{OH})_2$ (Ni^{2+}) to NiOOH (Ni^{3+}) at a given potential. The oxidation of glucose molecules to gluconolactone can be realized by NiOOH (Ni^{3+}), which also enables the recycling of $\text{Ni}(\text{OH})_2$ (Ni^{2+}) for glucose oxidation. When glucose is applied to an Ni electrode, the anodic peak current rises, while the cathodic peak current remains the same. Furthermore, metal and metal oxide nanomaterials can directly promote conjugation to bio/chemical molecules based on intrinsic surface properties. For example, noble metals are known for their strong bonds with sulfur-containing molecules, and metal oxides are known to bind with carboxyl groups. In addition, as catalytic oxidation efficiency is highly affected by the surface state of metals and metal oxides, most of the oxidation processes are conducted in alkaline solutions. Though this unusual condition can enhance catalytic/sensing performance, it is not practical for real, biological sample analyses.

3.1.2 Conducting polymers. Many kinds of conducting polymers such as polyaniline (PANI), poly(3,4-ethylene dioxthiophene) (PEDOT), and polypyrrole (PPy) are also considered as promising alternative materials for advanced electrochemical biosensors.^{89,92,112–116} The advantages of conducting polymers mainly include the ease of synthesis, a relatively high electronic/ionic conductivity, and the capability to form highly uniform arrays. The electrical conductivity of conducting polymers can be easily modulated by the doping degree. For example, when nonconductive polyaniline (PANI) is exposed to protonic acids such as HCl or camphor sulfonic acid, the nitrogen atoms on the imine group of the PANI backbone are protonated, where the pH-dependent degree of protonation results from its oxidation status.⁹² Through the complete protonation of the emeraldine base form, conductivity can be increased by up to 10 orders of magnitude by forming a delocalized polysemiquinone radical cation. Though this mechanism is reversible through manipulation of the environmental conditions, these polymers can be utilized as transducers and can monitor pH through electrical conductivity changes. Interestingly, in this case, many electrons along the polymer backbone did not change during the doping process; however, the energy levels of polymers were rearranged. Conversely, other conducting polymers such as PEDOT and PPy can be doped through a redox reaction.⁸⁹ As these polymers are also nonconductive under normal conditions and possess a positively charged monomer unit on their backbones, they are prone to attract negatively charged dopants, which results in charge balancing on the polymer backbone with improved electrical conduction. Additionally, conducting polymers with well-defined redox behavior can function as good electrode materials for electrochemical sensing systems. However, their poor mechanical stability in aqueous electrolyte solutions restrains further progress in real applications during long-term cycling.

3.1.3 Carbon-based materials. Carbon-based nanomaterials such as carbon nanotubes (CNTs), graphene, and its derivatives have been extensively studied and utilized for devel-

oping electrochemical sensing systems.^{117–122} CNTs and pristine graphene are known to have similar conducting properties based on their unique structural arrangements.^{123–125} The sp^2 carbon bond is perfectly arranged in a honeycomb lattice, which results in delocalization of π electrons across the planar plane. Carbon-based nanomaterials with oxygen-containing groups such as graphene derivatives, including graphene oxide (GO) and reduced graphene oxide (rGO), are less conductive than the pristine carbon-based materials as the electron delocalization is disturbed by the formation of sp^3 carbon bonds. Thus, higher oxygen to carbon ratios result in reduced electrical conductivity; however, this also known to be beneficial in some cases of electrochemical sensing systems. For example, oxygen-containing groups (e.g., carboxyl group) that are known as defects can promote the adhesion of primary amine-containing molecules or positively charged surfaces.¹²⁶ In addition, they can also serve as anchoring sites for metal nanoparticle formation for the development of nonenzymatic electrochemical sensing systems as well.^{127,128} Furthermore, doping with foreign atoms can also generate additional functionality for carbon-based materials.¹²³ For example, as nitrogen possesses five valence electrons and is similar in size to carbon, it is expected to form strong valence bonds with carbon atoms. Thus, nitrogen has been typically utilized for doping carbon-based materials, as the enhancement of conductivity could be expected based on the increased number of free charge carriers. In addition, nitrogen-doped carbon-based materials also exhibited excellent catalytic activity toward oxygen reduction reactions similar to metal nanoparticle decorations under certain conditions.¹²⁹ Furthermore, on carbon-based materials, a wide variety of aromatic group-containing bio/chemical molecules can noncovalently adsorb *via* strong π - π interactions, which can preconcentrate the analytes on the electrode surface. However, these excellent binding moieties provided by either pristine or oxidized carbon-based materials can promote surface fouling, which hampers the sensitivity and selectivity of electrochemical sensing systems as well.¹²⁶ Phenolic and amine compounds are particularly concerning, as they are universal in unprocessed biological samples.

3.2 Non-destructive, live-cell monitoring

One relatively simple study on non-destructive, live-cell monitoring was achieved by H_2O_2 monitoring. Shu *et al.* used MnO_2 nanosheets to cover a glass carbon electrode and electrochemically detected H_2O_2 [Fig. 5].⁹¹ The inclusion of the nanosheets resulted in a much greater reduction current under cyclic voltammetry, permitting amperometric detection of H_2O_2 concentrations as low as 5 μM with two linear ranges of 0.025–2 μM and 10–454 μM that were stable over at least 2 weeks and 100 scan cycles. The increased sensitivity was attributed to the catalytic reduction of H_2O_2 on the MnO_2 surface and porous nanostructure, leading to the high specific surface area. Moreover, the addition of uric acid, glucose, and ascorbic acid did not appreciably interfere with H_2O_2 amphoteric responses at -0.6 V or subsequent H_2O_2 addition and detection. To examine the applicability of the sensor in the detection of

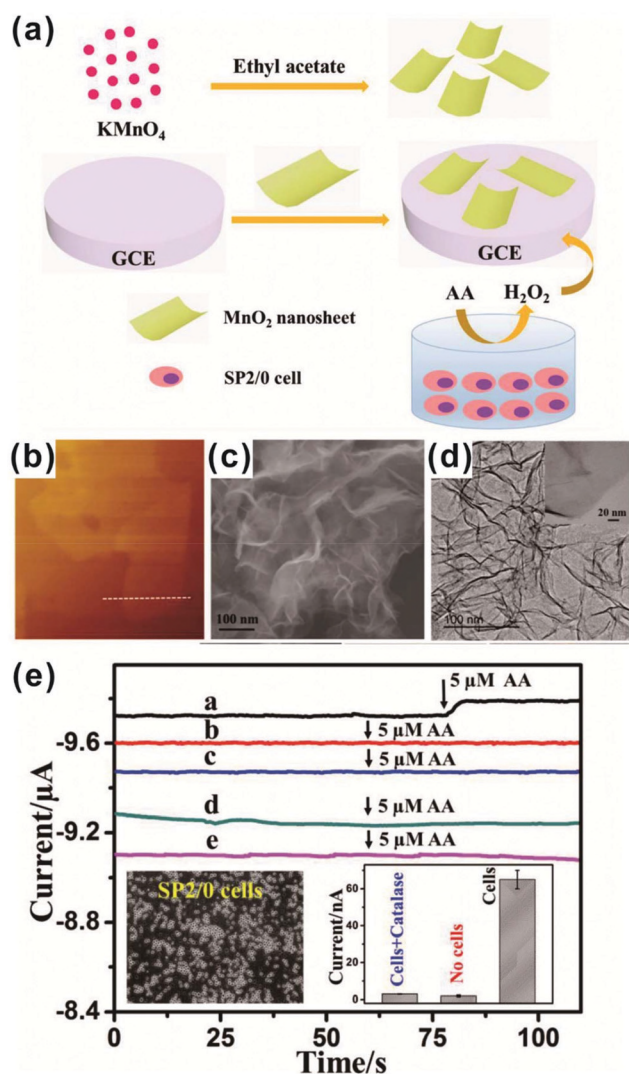


Fig. 5 Non-destructive, live-cell monitoring technique by H₂O₂ monitoring using MnO₂ nanosheets on a glassy carbon electrode (GCE). (a) Schematic illustration of H₂O₂ detection from SP2/0 cells. (b–d) Characterization of MnO₂ nanosheets. (e) Amperometric *i-t* curves of the response of the GCE with MnO₂ nanosheets (curve a) and without MnO₂ nanosheets (curve d) for the reduction of H₂O₂ released from SP2/0 cells. Amperometric *i-t* curves of the response of the GCE with MnO₂ nanosheets (curve b) and without MnO₂ nanosheets (curve e) in the absence of SP2/0 cells. Amperometric *i-t* curves of the response of the GCE with MnO₂ nanosheets (curve c) in the presence of SP2/0 cells and catalase. Reprinted with permission from ref. 91. © 2017 Elsevier B. V. All rights reserved.

H₂O₂ from live cells, Sp2/0 was cultured, suspended in PBS, and stimulated to release H₂O₂ *via* the addition of ascorbic acid. The detected signal was consistent with other reports (40 amol H₂O₂ per cell at 5 μM ascorbic acid for 10⁷ cells in 4 mL PBS) and was not present in no cell, no ascorbic acid, and catalase controls. This study is a prototypical example of how nanomaterials can be used to improve the electrochemical detection of biologically relevant molecules. In order to improve electrochemical detection performance, electrochemical sensing systems have naturally led to designs incor-

porating multiple metal species. Zhu and coworkers focused on growing PtW nano-cubes on MoS₂ nanosheets *via* thermal decomposition.³¹ The resulting hybrid nanocomposite was stable under ultrasonication and exhibited lower electron transfer resistance than the bare electrode surface or electrodes coated with MoS₂. Amperometric analysis at −0.25 V showed detection to be linear from 1 μM to 200 μM, with a detection limit of 5 nM H₂O₂. Interestingly, Pt on MoS₂, PtW on graphene, and PtW on WS₂ all showed nonspecific responses to ascorbic acid and uric acid, whereas the PtW/MoS₂ system showed negligible disruption and enhanced specificity. Further specificity tests using dopamine, NO₃[−], K⁺, 4-acetamidophenol, and hemoglobin showed that the PtW/MoS₂ sensor was unaffected and still capable of detecting further addition of H₂O₂. A cellular demonstration was conducted using 4T1 induced to release H₂O₂ *via* N-formyl methionyl leucyl phenylalanine (fMLP) stimulation. For this particular system, 2 × 10⁶ cells could be induced to release a maximum of 25 nM H₂O₂ after stimulation with 2.2 μM fMLP, which is in agreement with other reports. As further work is done to enhance sensor specificity and sensitivity, more work investigating hybrid systems will become more commonplace. Similarly, Hu *et al.* deposited platinum nanoparticles and graphene on a sacrificial 3D porous nickel foam.¹⁰⁰ The resulting Pt/graphene scaffold enabled significantly higher specific surface area, which increased the charge/mass-transfer rate, overall conductivity, and A375 cell adhesion on the interior and exterior surfaces, as compared to a 2D Pt/graphene nanosheet counterpart. Interestingly, the 3D graphene foam was found to have a more positive zeta potential than 2D graphene nanosheets. Reduced electrostatic repulsion between superoxide anions and 3D graphene foams, as compared to 2D nanosheets, was proposed as a reason why 3D foams displayed better electrochemical detection performance. Detection of superoxide anions (at 0.6 μM) was found to be unimpeded by ascorbic acid, K⁺, Na⁺, Ca²⁺, Cl[−], or NO₃[−] up to 6.5 μM. When A375 cells were induced to release superoxide anions *via* zymosan, 3D Pt/graphene foams again showed greater sensitivity than 2D nanosheet films seeded with an equivalent number of cells. This demonstration shows that 3D environments can not only support better cell function but also assist with electrochemical performance.

Although various electrochemical sensing systems have been extensively developed for monitoring soluble cell signaling components, it can be advantageous to develop a sensor that not only augments electrochemical detection but also enables direct monitoring of long-term live-cell growth. To this end, one group used WS₂ nanoflowers to coat carbon fiber electrochemical sensors for simultaneous H₂O₂ monitoring and cell culture.²⁹ The WS₂ nanosheets naturally adopted a flower-like structure after chemical vapor deposition, which dramatically increased the specific surface area and exposed edge sites. An amperometric detection limit of 2 nM H₂O₂ has been reported, which was not affected by the presence of superoxide anions or hypochlorite anions at a constant −0.25 V. To demonstrate the *in vitro* applications of this sensor system, RAW 264.7 macrophages, and primary rat hippocam-

pal neurons were cultured and induced to release H_2O_2 via *N*-formyl methionyl leucyl phenylalanine (fMLP) and epidermal growth factor (EGF), respectively. The authors demonstrated the detection of H_2O_2 from $0.3 \mu\text{M}$ fMLP and 500 ng ml^{-1} EGF from the respective test cells. Moreover, the EGF-induced signal can be attenuated by the addition of PD153035 (EGF receptor kinase domain inhibitor), wortmannin (phosphatidylinositol-3-OH kinase inhibitor), NSC23766 (NO synthase inhibitor), and apocynin (Nox inhibitor). This paper not only demonstrates an improvement in the field of metal-facilitated electrochemical detection (direct cell culture on sensors), but

also broadens the potential applications of such sensors to probing various cell–drug interactions as long as H_2O_2 is involved at some point in the interaction. In a more advanced manner, one communication of note from Lee *et al.* simultaneously investigated nanotopography to increase the efficacy of mesenchymal stem cell differentiation while enabling more sensitive electrochemical detection of key differentiation markers.²⁸ Arrays of graphene-coated gold nano-domes were fabricated with various widths and pitch sizes, and they served to modulate mechanotransduction signaling for osteogenesis. Dome widths of 400 nm were found to be the most conducive

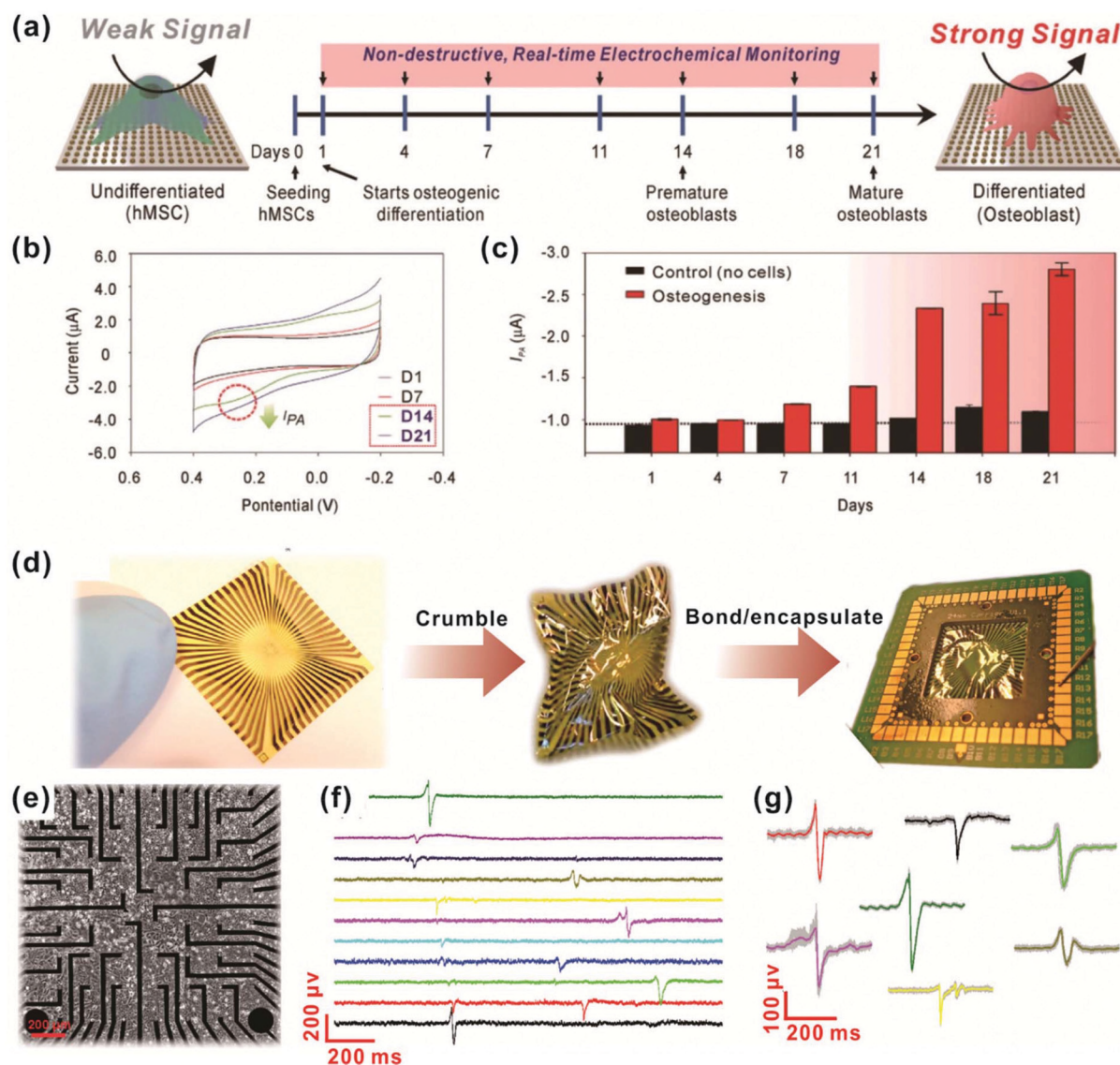


Fig. 6 Non-destructive, live-cell monitoring methods using electrochemical sensing (a–e) and electrophysiological sensing (d–g). (a) Schematic diagram of the electrochemical signal change during osteogenic differentiation of MSC on the nanoarray composed of gold and reduced graphene oxide. (b) Cyclic voltammogram of cultured MSC on the nanoarray from time-dependent monitoring (Day 0 to Day 21). (c) The cathodic peak currents of the MSC cultured nanoarray from day 1 to day 21. (d) Tested graphene microelectrodes for heart tissue recording. A flexible chip was crumbled to mechanical deformation, then soldered and encapsulated. (e) Picture of HL-1 cells seeded on graphene microelectrodes. (f) Time trace recordings of HL-1 cells on 11 different channels of graphene microelectrodes. (g) The variety of recorded action potential shapes from different HL-1 cells. (a)–(e) are reprinted with permission from ref. 28. © 2018 WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim, and (d)–(g) are reprinted with permission from ref. 101. © 2016 by the authors. Licensee MDPI, Basel, Switzerland.

to osteogenesis, as measured from alkaline phosphatase [ALP] and osteocalcin expression. To demonstrate the platform for nondestructive stem cell differentiation monitoring, the authors used voltammetry for live cell, *in situ* ALP assays. The graphene-coated nano-domes enabled higher sensitivity towards the electrochemically active ALP reaction product, due to a higher electron transfer rate. Moreover, this system was used to nondestructively monitor osteogenesis over 3 weeks with corresponding increases in ALP electrochemical activity as the culture time increased. This demonstrates the potential for the utilization of nanotopography and nanomaterials for multiple purposes, including the enhancement of stem cell differentiation and nondestructive monitoring [Fig. 6a–c].

Furthermore, while many biosensors are serviceable *in vitro*, many researchers are trying to integrate conventional electrodes based on glass and metals into living animals but the methods are complicated by severe mechanical mismatch. To develop biosensors capable of both *in vitro* and future *in vivo* sensing, Kireev *et al.* fabricated relatively flexible and soft polyimide substrates to detect cardiomyocyte electrophysiological function.¹⁰¹ Since action potential-based ion movement was of interest, the authors utilized electrical impedance spectroscopy to monitor cardiomyocyte function. The fabricated sensors could be crumpled and would still function to record electrical activity from HL-1 cardiomyocytes. Cells were found to be able to proliferate on the sensor and generate action potential waveforms that are similar to those found elsewhere in the literature. Different electrode channels recorded signals with a time delay, indicating that action potentials propagate through the cell monolayer with potential future applications in spatially monitoring where and how action potentials behave throughout a tissue. Dissected rat embryonic heart tissue was placed on the flexible sensor and found to have a signal-to-noise ratio of approximately 50. Moreover, the shape of the recorded action potentials shows typical P, Q, R, S, and T regions of normal electrocardiograms [Fig. 6d–g]. For *in vivo* biosensing to become more clinically relevant, additional efforts, such as those described here, will be needed to address tissue–substrate mechanical mismatch (Table 4).

Table 4 Nanomaterials for biosensing

| Material | Target of detection | Ref. |
|-----------------|---------------------------------|------|
| Inorganic metal | Hemisphere Au | 106 |
| | Pt-Nanocube on MoS ₂ | 31 |
| | MnO ₂ | 91 |
| | WS ₂ | 29 |
| Polymer | PANI | 92 |
| | Polyimide | 101 |
| | Pt/graphene nanosheet | 100 |
| Carbon-based | Graphene-Au | 28 |
| | Glucose | |
| | H ₂ O ₂ | |
| | H ₂ O ₂ | |
| | H ₂ O ₂ | |
| | Uric acid | |
| | Cholesterol | |
| | Triglycerides | |
| | Electrophysiological signal | |
| | Superoxide anion | |
| | Alkaline phosphatase | |

4. Concluding remarks and future perspectives

In this review, we summarized the recent advancements of nanoarrays for stem cell-based therapy. The combination of extracellular environment modulation and precise stem cell behavior and function monitoring is the focus of functional nanoarrays. Through this review, we believe that the ideal stem cell nanoarray should offer these distinct advantages: (1) biocompatibility, (2) nanotopographical cues capable of directing stem cell behaviors, and (3) enhanced sensitivity and selectivity to characterize cell fate and function non-destructively. In turn, the next-generation nanoarrays for cell monitoring and differentiation must leverage additional benefits from nanotechnology and materials science. To this end, an increasingly interdisciplinary knowledge base is required to integrate these fields for the successful development of a clinically relevant product. Thus, advanced approaches are needed to utilize functional nanoarrays for manipulating and monitoring stem cell behavior in real time, which will lead to the development of materials and methods for treating numerous diseases and disorders.

Looking forward, key perspectives on both fabrication and biological analysis should be achieved. One of the primary barriers to nanotopography applications is the limited active surface area generated by the current fabrication techniques. Considering the real therapeutic approaches, clinical usage is still restricted, due to the limited size. Thus, developing a controllable and cost-effective nanoarray synthesis method that can be easily scaled up is one urgent requirement. Additionally, biocompatible flexible substrates can further facilitate clinical translation as well. Furthermore, understanding how the spatial structural morphology of nanoarrays contributes to different aspects, including the interfacial properties that affect the interaction with cells and electrical/electrochemical performances such as electron transfer rates, should be considered. A three-dimensional culture environment is also an aspect for which more information on the stem cell-environment regulation is required to understand the resulting cell signaling and epigenetic interactions.

This review also showcased how functional nanomaterials could provide answers to challenges in the field of biological analysis when applied as transducers for electrical/electrochemical biosensors. Ideal biosensors need to detect low abundance analytes from a small volume of the original matrix in a rapid, accurate, and inexpensive manner while maintaining long-term stability under the complex physiological assay environment. The successful integration of nanomaterials for cell-based biological analysis approaches must maintain function under physiological conditions such as high ionic strength and body temperatures. While these challenges can be overcome with appropriate surface modifications, a balance needs to be achieved between appropriate surface chemistry and the desired functionality in biosensing under physiological conditions.

Furthermore, there is still much room to improve beyond the nanoelectrode-based approaches for stem cell-based therapy. For example, it should be possible to perform stimulation and monitoring of stem cell differentiation simultaneously. Instead of direct electrical stimulation, nanoelectrode arrays can also be used for the highly topical, cell-specific delivery system for cell-controlling molecules, such as proteins, DNA, and nanoparticles, while altering integrin and cadherin ligations through topographical effects as well. To this end, we envision that the application of nanoarrays in studies related to stem cell behavior and function will not only facilitate the implementation of stem cell therapies to treat degenerative diseases and injuries, but also drive stem cell therapy closer to clinical applications. This will ultimately lead to the development of methods, tools, drugs, and materials for treating numerous diseases and disorders. Thus, incorporating recent high-throughput screening approaches will likely be instrumental in the discovery of new drugs and materials and the design and development of novel materials to control and monitor stem cell fate.

Conflicts of interest

The authors do not have any conflicts of interest to declare.

Acknowledgements

W. C. acknowledges partial financial support from the NRF (2019R1A2C3002300) and (2016R1A6A1A03012845) funded by MSIP and the ME of Korea. K.-B. L. acknowledges partial financial support from NIH R21 (R21AR071101), NIH R01 (1R01DC016612, 3R01DC016612-01S1, and 5R01DC016612-02S1), New Jersey Commission on Spinal Cord Research (CSCR17IRG010 and CSCR16ERG019), and NSF (CBET-1803517). J.-H. L acknowledges partial financial support from the NRF (2019R1H1A1A01058779).

Notes and references

- 1 A. Ascherio and M. A. Schwarzschild, *Lancet Neurol.*, 2016, **15**, 1257–1272.
- 2 A. Trounson and N. D. DeWitt, *Nat. Rev. Mol. Cell Biol.*, 2016, **17**, 194–200.
- 3 I. J. Fox, G. Q. Daley, S. A. Goldman, J. Huard, T. J. Kamp and M. Trucco, *Science*, 2014, **345**, 1247391.
- 4 P. S. Knoepfler, *Adv. Drug Delivery Rev.*, 2015, **82–83**, 192–196.
- 5 J. L. Sternecker, P. Reinhardt and H. R. Scholer, *Nat. Rev. Genet.*, 2014, **15**, 625–639.
- 6 S. Dimmeler, S. Ding, T. A. Rando and A. Trounson, *Nat. Med.*, 2014, **20**, 814–821.
- 7 K. G. Chen, B. S. Mallon, R. D. McKay and P. G. Robey, *Cell Stem Cell*, 2014, **14**, 13–26.
- 8 Y. Shi, H. Inoue, J. C. Wu and S. Yamanaka, *Nat. Rev. Drug Discovery*, 2017, **16**, 115–130.
- 9 M. J. Dalby, N. Gadegaard and R. O. Oreffo, *Nat. Mater.*, 2014, **13**, 558–569.
- 10 X. Wang, S. Li, C. Yan, P. Liu and J. Ding, *Nano Lett.*, 2015, **15**, 1457–1467.
- 11 J. H. Tsui, K. Janebodin, N. Ieronimakis, D. M. P. Yama, H. S. Yang, R. Chavanachai, A. L. Hays, H. Lee, M. Reyes and D. H. Kim, *ACS Nano*, 2017, **11**, 11954–11968.
- 12 J. Yang, L. E. McNamara, N. Gadegaard, E. V. Alakpa, K. V. Burgess, R. M. Meek and M. J. Dalby, *ACS Nano*, 2014, **8**, 9941–9953.
- 13 K. H. Vining and D. J. Mooney, *Nat. Rev. Mol. Cell Biol.*, 2017, **18**, 728–742.
- 14 A. J. Engler, S. Sen, H. L. Sweeney and D. E. Discher, *Cell*, 2006, **126**, 677–689.
- 15 J. Du, X. Chen, X. Liang, G. Zhang, J. Xu, L. He, Q. Zhan, X. Q. Feng, S. Chien and C. Yang, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 9466–9471.
- 16 H. Lv, L. Li, M. Sun, Y. Zhang, L. Chen, Y. Rong and Y. Li, *Stem Cell Res. Ther.*, 2015, **6**, 103.
- 17 M. Brunner, N. Mandier, T. Gautier, G. Chevalier, A. S. Ribba, P. Guardiola, M. R. Block and D. Bouvard, *PLoS One*, 2018, **13**, e0196021.
- 18 Y. R. Shih, K. F. Tseng, H. Y. Lai, C. H. Lin and O. K. Lee, *J. Bone Miner. Res.*, 2011, **26**, 730–738.
- 19 C. D. Kegelman, D. E. Mason, J. H. Dawahare, D. J. Horan, G. D. Vigil, S. S. Howard, A. G. Robling, T. M. Bellido and J. D. Boerckel, *FASEB J.*, 2018, **32**, 2706–2721.
- 20 K. F. Wong, A. M. Liu, W. Hong, Z. Xu and J. M. Luk, *Oncotarget*, 2016, **7**, 77683–77695.
- 21 P. Y. Wang, H. Thissen and P. Kingshott, *Acta Biomater.*, 2016, **45**, 31–59.
- 22 E. Ngandu Mpoyi, M. Cantini, P. M. Reynolds, N. Gadegaard, M. J. Dalby and M. Salmerón-Sánchez, *ACS Nano*, 2016, **10**, 6638–6647.
- 23 M. Salmeron-Sanchez and M. J. Dalby, *Chem. Commun.*, 2016, **52**, 13327–13336.
- 24 C. Uhler and G. V. Shivashankar, *Nat. Rev. Mol. Cell Biol.*, 2017, **18**, 717–727.
- 25 S. Nemati, S. J. Kim, Y. M. Shin and H. Shin, *Nano Convergence*, 2019, **6**, 36.
- 26 H. S. Goodwin, A. R. Bicknese, S. N. Chien, B. D. Bogucki, C. O. Quinn and D. A. Wall, *Biol. Blood Marrow Transplant.*, 2001, **7**, 581–588.
- 27 T. H. Kim, C. H. Yea, S. T. Chueng, P. T. Yin, B. Conley, K. Dardir, Y. Pak, G. Y. Jung, J. W. Choi and K. B. Lee, *Adv. Mater.*, 2015, **27**, 6356–6362.
- 28 J. H. Lee, H. K. Choi, L. Yang, S. D. Chueng, J. W. Choi and K. B. Lee, *Adv. Mater.*, 2018, **30**, e1802762.
- 29 J. Tang, Y. Quan, Y. Zhang, M. Jiang, A. M. Al-Enizi, B. Kong, T. An, W. Wang, L. Xia and X. Gong, *Nanoscale*, 2016, **8**, 5786–5792.
- 30 J. H. Lee, J. H. Choi, S. D. Chueng, T. Pongkulapa, L. Yang, H. Y. Cho, J. W. Choi and K. B. Lee, *ACS Nano*, 2019, **13**, 8793–8803.
- 31 L. Zhu, Y. Zhang, P. Xu, W. Wen, X. Li and J. Xu, *Biosens. Bioelectron.*, 2016, **80**, 601–606.

- 32 P. K. Nguyen, J. W. Rhee and J. C. Wu, *JAMA Cardiol.*, 2016, **1**, 831–841.
- 33 B. Mead, A. Logan, M. Berry, W. Leadbeater and B. A. Scheven, *Stem Cells*, 2017, **35**, 61–67.
- 34 C. Xue, X. Shi, X. Fang, H. Tao, H. Zhu, F. Yu, X. Ding, M. Liu, F. Fang, F. Yang, Z. Wei, T. Chen, Z. Wang, G. Wang, X. Cheng, J. Wei, Y. Lin, K. Deng, X. Wang and H. Xin, *ACS Appl. Mater. Interfaces*, 2016, **8**, 8393–8400.
- 35 L. Luo, Q. P. Jiang, G. H. Qin, K. Zhao, G. F. Du, H. Wang and H. Y. Zhao, *Sens. Actuators, B*, 2015, **218**, 205–214.
- 36 M. Zhang, S. Wang, Z. L. Li, C. W. Liu, R. Miao, G. He, M. Zhao, J. Xue, Z. Y. Xia, Y. Q. Wang, Z. Q. Sun and J. G. Lv, *RSC Adv.*, 2019, **9**, 3479–3485.
- 37 K. Zhao, G. F. Du, L. Luo, G. H. Qin, Q. P. Jiang, Y. J. Liu and H. Y. Zhao, *CrystEngComm*, 2015, **17**, 2030–2040.
- 38 Y. Zhou, S. Q. Xi, X. G. Yang and H. J. Wu, *J. Solid State Chem.*, 2019, **270**, 398–406.
- 39 F. Gao, G. H. Qin, Y. H. Li, Q. P. Jiang, L. Luo, K. Zhao, Y. J. Liu and H. Y. Zhao, *RSC Adv.*, 2016, **6**, 10298–10310.
- 40 B. Cook, Q. F. Liu, J. W. Liu, M. G. Gong, D. Ewing, M. Casper, A. Stramel and J. D. Wu, *J. Mater. Chem. C*, 2017, **5**, 10087–10093.
- 41 S. B. Wang, Y. C. Wu, R. Miao, M. W. Zhang, X. X. Lu, B. Zhang, A. Kinstler, Z. Y. Ren, Y. B. Guo, T. F. Lu, S. L. Suib and P. X. Gao, *CrystEngComm*, 2017, **19**, 5128–5136.
- 42 C. Young, J. Wang, J. Kim, Y. Sugahara, J. Henzie and Y. Yamauchi, *Chem. Mater.*, 2018, **30**, 3379–3386.
- 43 L. Kang, S. Zhang, Q. Li and J. Zhang, *J. Am. Chem. Soc.*, 2016, **138**, 6727–6730.
- 44 X. Zhang, Y. Zhang, B. B. Yu, X. L. Yin, W. J. Jiang, Y. Jiang, J. S. Hu and L. J. Wan, *J. Mater. Chem. A*, 2015, **3**, 19277–19281.
- 45 C. Soci, A. Zhang, B. Xiang, S. A. Dayeh, D. P. Aplin, J. Park, X. Y. Bao, Y. H. Lo and D. Wang, *Nano Lett.*, 2007, **7**, 1003–1009.
- 46 L. Xu, X. Li, Z. Zhan, L. Wang, S. Feng, X. Chai, W. Lu, J. Shen, Z. Weng and J. Sun, *ACS Appl. Mater. Interfaces*, 2015, **7**, 20264–20271.
- 47 Y. Chen, X. Zhang and Z. Xie, *ACS Nano*, 2015, **9**, 8054–8063.
- 48 H. Zhou, X. P. Zou and Y. R. Zhang, *Electrochim. Acta*, 2016, **192**, 259–267.
- 49 M. Yu, Y. X. Ma, J. H. Liu and S. M. Li, *Carbon*, 2015, **87**, 98–105.
- 50 J. Dong, L. Ren, Y. Zhang, X. Cui, P. Hu and J. Xu, *Talanta*, 2015, **132**, 719–726.
- 51 N. Tsyntsar, S. Silkin, H. Cesiulis, M. Guerrero, E. Pellicer and J. Sort, *Electrochim. Acta*, 2016, **188**, 589–601.
- 52 X. Wang, K. Ye, H. Y. Zhang, X. K. Ma, K. Zhu, K. Cheng, G. L. Wang and D. X. Cao, *Int. J. Hydrogen Energy*, 2017, **42**, 15044–15053.
- 53 D. Gioia and I. G. Casella, *Sens. Actuators, B*, 2016, **237**, 400–407.
- 54 C. R. Martin, *Science*, 1994, **266**, 1961–1966.
- 55 C. Zhao, Y. Zhu, L. Chen, S. Zhou, Y. Su, X. Ji, A. Chen, X. Gui, Z. Tang and Z. Liu, *Nanoscale*, 2018, **10**, 16278–16283.
- 56 X. K. Ma, K. Ye, G. Wang, M. Y. Duan, K. Cheng, G. L. Wang and D. X. Cao, *Appl. Surf. Sci.*, 2017, **414**, 353–360.
- 57 Y. Xu, M. Zhou, L. Y. Wen, C. L. Wang, H. P. Zhao, Y. Mi, L. Y. Liang, Q. Fu, M. H. Wu and Y. Lei, *Chem. Mater.*, 2015, **27**, 4274–4280.
- 58 X. F. Guan, D. Chen, Z. Y. Quan, F. X. Jiang, C. H. Deng, G. A. Gehring and X. H. Xu, *Nanoscale Res. Lett.*, 2015, **10**, 2419.
- 59 Q. Hao, H. Huang, X. Fan, X. Hou, Y. Yin, W. Li, L. Si, H. Nan, H. Wang, Y. Mei, T. Qiu and P. K. Chu, *Nanotechnology*, 2017, **28**, 105301.
- 60 H. Robatjazi, S. M. Bahaeddin, L. H. Macfarlan, S. D. Fu and I. Thomann, *Chem. Mater.*, 2016, **28**, 4546–4553.
- 61 L. Wen, R. Xu, Y. Mi and Y. Lei, *Nat. Nanotechnol.*, 2017, **12**, 244–250.
- 62 L. Wen, R. Xu, C. Cui, W. Tang, Y. Mi, X. Lu, Z. Zeng, S. L. Suib, P. X. Gao and Y. Lei, *Nano Lett.*, 2018, **18**, 4914–4921.
- 63 G. Abagnale, M. Steger, V. H. Nguyen, N. Hersch, A. Sechi, S. Joussen, B. Denecke, R. Merkel, B. Hoffmann, A. Dreser, U. Schnakenberg, A. Gillner and W. Wagner, *Biomaterials*, 2015, **61**, 316–326.
- 64 C. Yang, M. W. Tibbitt, L. Basta and K. S. Anseth, *Nat. Mater.*, 2014, **13**, 645–652.
- 65 T. D. Bornes, N. M. Jomha, A. Mulet-Sierra and A. B. Adesida, *Tissue Eng., Part C*, 2016, **22**, 208–220.
- 66 E. B. Hunziker, T. M. Quinn and H. J. Hauselmann, *Osteoarthritis Cartilage*, 2002, **10**, 564–572.
- 67 S. Moghadasi Boroujeni, S. Mashayekhan, S. Vakilian, A. Ardeshtyrlajimi and M. Soleimani, *J. Biomed. Mater. Res., Part A*, 2016, **104**, 1610–1621.
- 68 R. Li, J. Xu, D. S. H. Wong, J. Li, P. Zhao and L. Bian, *Biomaterials*, 2017, **145**, 33–43.
- 69 M. S. Lee and H. S. Yang, *Biotechnol. J.*, 2017, **12**, 1700309.
- 70 D. Carson, M. Hnilova, X. Yang, C. L. Nemeth, J. H. Tsui, A. S. Smith, A. Jiao, M. Regnier, C. E. Murry, C. Tamerler and D. H. Kim, *ACS Appl. Mater. Interfaces*, 2016, **8**, 21923–21932.
- 71 T. Jiang, D. Kai, S. Liu, X. Huang, S. Heng, J. Zhao, B. Q. Y. Chan, X. J. Loh, Y. Zhu, C. Mao and L. Zheng, *Biomaterials*, 2018, **178**, 281–292.
- 72 Y. Wu, Z. Yang, J. B. Law, A. Y. He, A. A. Abbas, V. Denslin, T. Kamarul, J. H. Hui and E. H. Lee, *Tissue Eng., Part A*, 2017, **23**, 43–54.
- 73 C. Zhao, X. Wang, L. Gao, L. Jing, Q. Zhou and J. Chang, *Acta Biomater.*, 2018, **73**, 509–521.
- 74 W. Qian, L. Gong, X. Cui, Z. Zhang, A. Bajpai, C. Liu, A. B. Castillo, J. C. M. Teo and W. Chen, *ACS Appl. Mater. Interfaces*, 2017, **9**, 41794–41806.
- 75 R. K. Assoian, N. D. Bade, C. V. Cameron and K. J. Stebe, *Open Biol.*, 2019, **9**, 190155.

- 76 A. J. Sophia Fox, A. Bedi and S. A. Rodeo, *Sports Health*, 2009, **1**, 461–468.
- 77 N. D. Leipzig and M. S. Shoichet, *Biomaterials*, 2009, **30**, 6867–6878.
- 78 S. Shah, P. T. Yin, T. M. Uehara, S. T. Chueng, L. Yang and K. B. Lee, *Adv. Mater.*, 2014, **26**, 3673–3680.
- 79 S. Ankam, C. K. Lim and E. K. Yim, *Biomaterials*, 2015, **47**, 20–28.
- 80 S. Ankam, B. K. K. Teo, G. Pohan, S. W. L. Ho, C. K. Lim and E. K. F. Yim, *Front. Bioeng. Biotechnol.*, 2018, **6**, 69.
- 81 L. Song, K. Wang, Y. Li and Y. Yang, *Colloids Surf., B*, 2016, **148**, 49–58.
- 82 K. K. B. Tan, W. W. M. Lim, C. Chai, M. Kukumberg, K. L. Lim, E. L. K. Goh and E. K. F. Yim, *Sci. Rep.*, 2018, **8**, 9567.
- 83 W. Chen, S. Han, W. Qian, S. Weng, H. Yang, Y. Sun, L. G. Villa-Diaz, P. H. Krebsbach and J. Fu, *Nanoscale*, 2018, **10**, 3556–3565.
- 84 A. J. Keung, E. M. de Juan-Pardo, D. V. Schaffer and S. Kumar, *Stem Cells*, 2011, **29**, 1886–1897.
- 85 R. G. Ireland and C. A. Simmons, *Stem Cells*, 2015, **33**, 3187–3196.
- 86 I. G. Kim, C. H. Gil, J. Seo, S. J. Park, R. Subbiah, T. H. Jung, J. S. Kim, Y. H. Jeong, H. M. Chung, J. H. Lee, M. R. Lee, S. H. Moon and K. Park, *Biomaterials*, 2018, **150**, 100–111.
- 87 H. Gerardo, A. Lima, J. Carvalho, J. R. D. Ramos, S. Couceiro, R. D. M. Travasso, R. Pires das Neves and M. Graos, *Sci. Rep.*, 2019, **9**, 9086.
- 88 G. Abagnale, A. Sechi, M. Steger, Q. Zhou, C. C. Kuo, G. Aydin, C. Schalla, G. Muller-Newen, M. Zenke, I. G. Costa, P. van Rijn, A. Gillner and W. Wagner, *Stem Cell Rep.*, 2017, **9**, 654–666.
- 89 K. Shavanova, Y. Bakakina, I. Burkova, I. Shteplyuk, R. Viter, A. Ubelis, V. Beni, N. Starodub, R. Yakimova and V. Khranovskyy, *Sensors*, 2016, **16**, 223.
- 90 J. M. Moon, N. Thapliyal, K. K. Hussain, R. N. Goyal and Y. B. Shim, *Biosens. Bioelectron.*, 2018, **102**, 540–552.
- 91 Y. Shu, J. Xu, J. Y. Chen, Q. Xu, X. Xiao, D. Q. Jin, H. Pang and X. Y. Hu, *Sens. Actuators, B*, 2017, **252**, 72–78.
- 92 L. Li, Y. Wang, L. Pan, Y. Shi, W. Cheng, Y. Shi and G. Yu, *Nano Lett.*, 2015, **15**, 1146–1151.
- 93 F. Hempel, J. K. Law, T. C. Nguyen, W. Munief, X. Lu, V. Pachauri, A. Susloparova, X. T. Vu and S. Ingebrandt, *Biosens. Bioelectron.*, 2017, **93**, 132–138.
- 94 H. Liu, L. Weng and C. Yang, *Microchim. Acta*, 2017, **184**, 1267–1283.
- 95 B. Nam, T.-K. Ko, S.-K. Hyun and C. Lee, *Nano Convergence*, 2019, **6**, 40.
- 96 M. Majdinasab, K. Mitsubayashi and J. L. Marty, *Trends Biotechnol.*, 2019, **37**, 898–915.
- 97 W. A. El-Said, T.-H. Kim, H. Kim and J.-W. Choi, *PLoS One*, 2011, **6**, e15836.
- 98 D. Grieshaber, R. MacKenzie, J. Voros and E. Reimhult, *Sensors*, 2008, **8**, 1400–1458.
- 99 W. A. El-Said, J. Yoon and J. W. Choi, *Nano Convergence*, 2018, **5**, 11.
- 100 F. X. Hu, C. X. Guo, H. B. Yang, Z. Z. Shi, M. Wang, Y. H. Xue, L. Zhu, T. Chen, L. M. Dai and C. M. Li, *Sens. Actuators, B*, 2019, **287**, 209–217.
- 101 D. Kireev, S. Seyock, M. Ernst, V. Maybeck, B. Wolfrum and A. Offenhausser, *Biosensors*, 2016, **7**, 1.
- 102 D. Guo, L. F. Lai, A. M. Cao, H. K. Liu, S. X. Dou and J. M. Ma, *RSC Adv.*, 2015, **5**, 55856–55869.
- 103 Y. Wang, X. Shan, H. Wang, S. Wang and N. Tao, *J. Am. Chem. Soc.*, 2017, **139**, 1376–1379.
- 104 N. Yusoff, A. Pandikumar, R. Ramaraj, H. N. Lim and N. M. Huang, *Microchim. Acta*, 2015, **182**, 2091–2114.
- 105 A. T. E. Vilian, W. Kim, B. Park, S. Y. Oh, T. Kim, Y. S. Huh, C. K. Hwangbo and Y. K. Han, *Biosens. Bioelectron.*, 2019, **142**, 111549.
- 106 C. W. Hsu, F. C. Su, P. Y. Peng, H. T. Young, S. Liao and G. J. Wang, *Sens. Actuators, B*, 2016, **230**, 559–565.
- 107 M. Fleischmann, K. Korinek and D. Pletcher, *J. Electroanal. Chem. Interfacial Electrochem.*, 1971, **31**, 39–49.
- 108 H. Zhu, L. Li, W. Zhou, Z. P. Shao and X. J. Chen, *J. Mater. Chem. B*, 2016, **4**, 7333–7349.
- 109 X. H. Niu, X. Li, J. M. Pan, Y. F. He, F. X. Qiu and Y. S. Yan, *RSC Adv.*, 2016, **6**, 84893–84905.
- 110 F. Y. Xie, T. T. Liu, L. S. Xie, X. P. Sun and Y. L. Luo, *Sens. Actuators, B*, 2018, **255**, 2794–2799.
- 111 S. Y. Tee, C. P. Teng and E. Ye, *Mater. Sci. Eng., C*, 2017, **70**, 1018–1030.
- 112 A. Valiūnienė, A. I. Rekertaitė, A. Ramanavičienė, L. Mikoliūnaitė and A. Ramanavičius, *Colloids Surf., A*, 2017, **532**, 165–171.
- 113 R. Jain, N. Jadon and A. Pawaiya, *TrAC, Trends Anal. Chem.*, 2017, **97**, 363–373.
- 114 G. Xu, Z. A. Jarjes, V. Desprez, P. A. Kilmartin and J. Travas-Sejdic, *Biosens. Bioelectron.*, 2018, **107**, 184–191.
- 115 G. Wang, R. Han, X. Su, Y. Li, G. Xu and X. Luo, *Biosens. Bioelectron.*, 2017, **92**, 396–401.
- 116 Z. Chao, S. Govindaraju, K. Giribabu, Y. S. Huh and K. Yun, *Sens. Actuators, B*, 2017, **252**, 616–623.
- 117 F. Tan, L. C. Cong, X. N. Li, Q. Zhao, H. X. Zhao, X. Quan and J. W. Chen, *Sens. Actuators, B*, 2016, **233**, 599–606.
- 118 P. Bollella, G. Fusco, C. Tortolini, G. Sanzo, G. Favero, L. Gorton and R. Antiochia, *Biosens. Bioelectron.*, 2017, **89**, 152–166.
- 119 H. Lee, T. K. Choi, Y. B. Lee, H. R. Cho, R. Ghaffari, L. Wang, H. J. Choi, T. D. Chung, N. Lu, T. Hyeon, S. H. Choi and D. H. Kim, *Nat. Nanotechnol.*, 2016, **11**, 566–572.
- 120 E. B. Bahadır and M. K. Sezgentürk, *TrAC, Trends Anal. Chem.*, 2016, **76**, 1–14.
- 121 Y. Song, Y. Luo, C. Zhu, H. Li, D. Du and Y. Lin, *Biosens. Bioelectron.*, 2016, **76**, 195–212.
- 122 J. A. Hondred, J. C. Breger, N. J. Alves, S. A. Trammell, S. A. Walper, I. L. Medintz and J. C. Claussen, *ACS Appl. Mater. Interfaces*, 2018, **10**, 11125–11134.

- 123 T. H. Kim, D. Lee and J. W. Choi, *Biosens. Bioelectron.*, 2017, **94**, 485–499.
- 124 J. H. Lee, S. J. Park and J. W. Choi, *Nanomaterials*, 2019, **9**, 297.
- 125 M. Pumera, *Mater. Today*, 2011, **14**, 308–315.
- 126 L. Vicarelli, S. J. Heerema, C. Dekker and H. W. Zandbergen, *ACS Nano*, 2015, **9**, 3428–3435.
- 127 W. Wei, G. Wang, S. Yang, X. Feng and K. Mullen, *J. Am. Chem. Soc.*, 2015, **137**, 5576–5581.
- 128 J. Ju and W. Chen, *Anal. Chem.*, 2015, **87**, 1903–1910.
- 129 Y. Tang, B. L. Allen, D. R. Kauffman and A. Star, *J. Am. Chem. Soc.*, 2009, **131**, 13200–13201.