

15                   **Imaging and control of critical fluctuations in two-dimensional magnets**

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30 **Abstract:**

31 Strong magnetization fluctuations are expected near the thermodynamic critical point of a  
32 continuous magnetic phase transition. Such critical fluctuations are highly correlated and in  
33 principle can occur at any time- and length-scales<sup>1</sup>; they govern critical phenomena and  
34 potentially can drive new phases<sup>2,3</sup>. Although critical phenomena in magnetic materials have  
35 been studied using neutron scattering, magnetic AC susceptibility and other techniques<sup>4-6</sup>, direct  
36 real-time imaging of critical magnetization fluctuations remains elusive. Here we develop a fast  
37 and sensitive magneto-optical imaging microscope to achieve wide-field, real-time monitoring of  
38 critical magnetization fluctuations in single-layer ferromagnetic insulator CrBr<sub>3</sub>. We track the  
39 critical phenomena directly from the fluctuation correlations and observe both slowing-down  
40 dynamics and enhanced correlation length. Through real-time feedback control of the critical  
41 fluctuations, we further achieve switching of magnetic states solely by electrostatic gating. The  
42 ability to directly image and control critical fluctuations in 2D magnets opens up exciting  
43 opportunities to explore critical phenomena and develop applications in nanoscale engines and  
44 information science.

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47 Two-dimensional (2D) van der Waals magnets have attracted much recent interest<sup>2,3,7-19</sup>. As  
48 atomically-thin layers, they have been incorporated into van der Waals heterostructures and  
49 device architectures to enable novel functionalities, such as gate-controllable magnetism and  
50 spin-filter tunneling<sup>9-14</sup>. The 2D nature of layered magnets also leads to distinct physical  
51 properties from that of their bulk counterparts, ranging from stacking-dependent magnetism to  
52 topological spin excitations<sup>15-19</sup>. Of particular interest are magnetization fluctuations near a  
53 thermodynamic critical point<sup>2,3</sup>, which are crucial to understanding magnetism in two dimensions  
54 but remain largely unexplored experimentally. In three dimensions the phase space for thermal  
55 fluctuations to become critical is small and difficult to access according to the Ginzburg  
56 criterion<sup>1,20</sup>. In one-dimension fluctuations are so strong that magnetic long-range order is  
57 typically destroyed<sup>21</sup>. 2D layered magnets, in which fluctuations and long-range order reach a  
58 good balance, are therefore ideal to access and harness critical magnetization fluctuations.

59 Here we demonstrate real-time imaging of critical magnetization fluctuations and direct  
60 determination of their temporal and spatial correlations in single-layer ferromagnetic insulator  
61  $\text{CrBr}_3$  by a fast and sensitive magneto-optical imaging technique. We observe macroscopic  
62 spatial correlations and orders-of-magnitude change in the correlation time within around 0.5 K  
63 of the Curie temperature  $T_c$ . The observed extreme sensitivity of magnetization fluctuations to  
64 environment is different from non-critical thermal fluctuations in minuscule systems, and enables  
65 unprecedented flexibility for control. We are thereby able to switch the magnetic state of 2D  
66  $\text{CrBr}_3$  in a non-volatile, magnetic-field-free and current-free manner by harnessing critical  
67 fluctuations as a stochastic driving force, i.e. by toggling the critical fluctuations based on a real-  
68 time measurement of the magnetic state.

69 Being stochastic in nature, direct observation of critical fluctuations in real-time for a large  
70 sample area is challenging. We have developed a real-time magnetic circular dichroism (MCD)  
71 imaging technique that combines high temporal (up to 100 frames per second) and spatial ( $\sim 600$   
72 nm) resolution with high sensitivity, which allows us to monitor the magnetization fluctuations  
73 in a single-layer ferromagnet  $\text{CrBr}_3$ . Figure 1a shows an illustration of our experimental setup. It  
74 enhances the optical contrast of MCD through polarization control (see Supplementary Note 2  
75 and Methods). The concept of polarization control is well known in polarization microscopy, but  
76 the enhancement of the optical contrast is limited by the low polarization extinction ratio, which  
77 is typically several hundred due to polarization distortion from high numerical-aperture (NA)  
78 objectives<sup>22,23</sup>. The key improvement of our technique is to separate the effective NA for  
79 illumination and imaging: A long-focal-length lens L1 focuses the illumination beam roughly at  
80 the back aperture of an objective, minimizing the effective NA for illumination and thus  
81 polarization distortion. Similarly, specular-reflected light from a featureless substrate also  
82 maintains high polarization purity. However, scattered light from features on the sample is  
83 collected with the large effective NA of the objective. The spatial resolution is thus not  
84 compromised, as there is no physical aperture limiting the NA in the collection path. We are  
85 therefore able to achieve simultaneously a high extinction ratio ( $> 3 \times 10^4$ ) and high spatial  
86 resolution ( $\sim 600$  nm) over the entire field of view. In this configuration, the optical contrast is  
87 approximately proportional to the MCD signal (and therefore also the out-of-plane magnetization  
88 of the sample), but is enhanced by more than 100 times compared to the MCD signal without  
89 polarization control. In addition, because the MCD signal is determined by the dielectric function  
90 of the sample and the local field factor of the environment, we can optimize the substrate to  
91 further improve the optical contrast by a few times (see Supplementary Note 3).

92       Figure 1b shows an optical microscopy image of a monolayer CrBr<sub>3</sub> sample S1 (inside the  
93       white dashed box), which is encapsulated with hexagonal boron nitride (hBN) on both sides. It  
94       shows a giant magneto-optical contrast of  $\pm 60\%$  for the two remanent magnetization states at 18  
95       K (Fig. 1c and 1d), which are prepared by cooling the sample under an out-of-plane magnetic  
96       field of different polarities. Here the optical contrast is evaluated by using reflection from the  
97       CrBr<sub>3</sub> sample above  $T_c$  as reference. A nearby CrBr<sub>3</sub> flake (inside the green dashed box) can also  
98       be seen, but will not be focused on because of its smaller size. We first characterize the CrBr<sub>3</sub>  
99       monolayer (averaged over a 3x3  $\mu\text{m}$  area in the center) away from  $T_c$ . The magnetization against  
100       magnetic field shows a hysteresis loop below  $T_c$  and paramagnetic behavior above  $T_c$  (Fig. 1e).  
101       We extract the remanent magnetization from the data below  $T_c$  (Fig. 1h) and the DC magnetic  
102       susceptibility  $\chi$  from the data above  $T_c$  (red circles, Fig. 1f). The remanent magnetization  
103       decreases when temperature  $T$  approaches  $T_c$  from below; and  $\chi$  increases dramatically when  $T$   
104       approaches  $T_c$  from above. In contrast, a thin bulk CrBr<sub>3</sub> flake ( $\sim 10$  nm thick) shows a much  
105       weaker temperature dependence in susceptibility above  $T_c$  and no remanent magnetization below  
106        $T_c$  (black squares, Fig. 1f). The lack of remanent magnetization is presumably due to domains or  
107       antiferromagnetic ordering from the interlayer exchange and/or dipole-dipole interaction<sup>7,8</sup>. We  
108       determine  $T_c$  by fitting the temperature-dependent susceptibility to a critical scaling law  $\chi \sim (T -$   
109        $T_c)^{-\gamma}$  (Fig. 1g). We obtain  $T_c \approx 22.3$  K and 28 K, and critical exponent  $\gamma \approx 2.4$  and 1.2 for the  
110       monolayer and thin bulk samples, respectively. The value of  $T_c$  for the monolayer determined  
111       from susceptibility is consistent with the temperature dependence of the magnetization amplitude,  
112       which shows a sharp drop near 22.3 K (Fig. 1h). The critical exponent  $\gamma$  of the thin bulk matches  
113       well with previous results for bulk CrBr<sub>3</sub>, and is close to the mean-field value ( $\gamma = 1$ )<sup>24</sup>. On the  
114       other hand, the critical exponent for monolayer CrBr<sub>3</sub> is between the predictions of a 2D Ising ( $\gamma$

115 = 1.75) and 2D Heisenberg ( $\gamma = 3$ ) model<sup>25,26</sup>. This can potentially be understood from a recent  
116 theoretical proposal that 2D CrBr<sub>3</sub> is described by a 2D XXZ model with anisotropic exchange  
117 interaction<sup>17</sup>, and shows a crossover behavior between the Ising and Heisenberg model. Although  
118 the exact model to describe 2D CrBr<sub>3</sub> is still under debate<sup>17-19</sup>, our observation demonstrates its  
119 distinct behavior from that of the bulk, and substantial deviation from the mean-field description  
120 due to enhanced fluctuations in 2D.

121 As temperature further approaches  $T_c$  (22.3K) from below, the magnetization in monolayer  
122 CrBr<sub>3</sub> starts to fluctuate spontaneously under zero external magnetic field. Such fluctuations are  
123 not observed in the bulk sample under the same experimental conditions (see supplementary  
124 movie 1). We analyze both the temporal and spatial dependences of the magnetic fluctuations in  
125 monolayer CrBr<sub>3</sub>. Figure 2a summarizes the fluctuation amplitude map of the magnetization,  
126  $\delta M(\mathbf{r}) = \sqrt{\langle M(\mathbf{r}, t)^2 \rangle - \langle M(\mathbf{r}, t) \rangle^2}$ , at representative temperatures (see supplementary movie 2-  
127 13 for the real-time magnetization fluctuations). Here  $M(\mathbf{r}, t)$  is the out-of-plane magnetization  
128 at a given point  $(\mathbf{r}, t)$  in space and time, and the time average  $\langle \dots \rangle$  is equivalent to the ensemble  
129 average by assuming ergodicity (see Methods). The magnetization fluctuations first emerge at  
130 the corners of the monolayer flake at 21.6 K, quickly expand into the center with increasing  
131 temperature, remain large until  $T_c$ , then rapidly diminish above  $T_c$ . Figure 2c shows the  
132 magnetization dynamics at a fixed location P1 (green circle in Fig. 2a) for varying temperatures.  
133 The measured magnetization fluctuation dynamics and their temperature dependence resemble  
134 the result of a Monte Carlo simulation for a 2D Ising model<sup>27</sup>. Particularly, in the vicinity of  $T_c$ ,  
135 the magnetization shows random values between that of the fully spin-up state (referred to as  
136 state “1”) and the fully spin-down state (state “0”). This strongly supports that the observed  
137 magnetization fluctuations are critical fluctuations. We have also carried out several control

138 experiments to exclude potential experimental artifacts for the observed magnetization  
139 fluctuations (see Supplementary Note 8). Particularly, we have verified that the probe light has  
140 no effect on the fluctuations except for a systematic temperature shift of  $\sim 0.05\text{K}$ .

141 The central quantity that describes the critical fluctuations is the fluctuation correlation  
142 function between magnetizations at point  $\mathbf{r}_1$  and  $\mathbf{r}_2$  separated by time  $\Delta t$  (Ref.<sup>1</sup>)

143 
$$C(\mathbf{r}_1, \mathbf{r}_2, \Delta t) = \langle M(\mathbf{r}_1, t)M(\mathbf{r}_2, t + \Delta t) \rangle - \langle M(\mathbf{r}_1, t) \rangle \langle M(\mathbf{r}_2, t + \Delta t) \rangle. \quad (1)$$

144 The temporal correlation at a given point,  $C(\mathbf{r}_1 = \mathbf{r}_2, \Delta t)$ , contains direct information of the  
145 dynamics in the critical regime. Figure 2d shows the temporal correlation function at position P1  
146 at varying temperatures. It is well described by  $e^{-\Delta t/\tau}$ , where  $\tau$  is the correlation decay time. The  
147 fluctuations are substantially slowed down at 21.8 K (green curve in Fig. 2c), leading to a  
148 marked increase in  $\tau$ . Figure 2e summarizes the correlation decay time as a function of  
149 temperature. The significant slowing-down (up to a second) is a consequence of the enhanced  
150 critical magnetic fluctuations in 2D and has been independently confirmed by AC susceptibility  
151 measurements (see Methods and Supplementary Note 7 for more discussions). The peak of the  
152 correlation decay time, however, occurs at about 0.5 K below the average  $T_c$  of the sample  
153 (vertical dashed line). A more complete picture is obtained by examining the spatial dependence  
154 of the correlation decay time (Fig. 2b). The correlation decay time shows orders-of-magnitude  
155 change across the sample at a fixed temperature. It is peaked at slightly different temperatures for  
156 different locations. The slowing-down dynamics is in accord with the large fluctuation amplitude  
157 shown in Fig. 2a, both occurring within a narrow temperature range ( $\pm 0.5\text{ K}$ ) of  $T_c$  and both  
158 showing strong spatial inhomogeneity. Potential sources for inhomogeneity include the physical  
159 boundary, spatial variations in the carrier doping density (which changes  $T_c$  as shown in Fig. 4),

160 local strain, etc. The extreme sensitivity of magnetic fluctuations to perturbations in the vicinity  
161 of the critical point further supports their critical nature.

162 The critical behavior of the magnetic fluctuations and the spatial inhomogeneity effects are  
163 also clearly seen in the spatial correlation function  $C(\mathbf{r}_1, \mathbf{r}_2, \Delta t = 0)$ . Figure 3a shows  $C(\mathbf{r}_1, \mathbf{r}_2)$   
164 at several representative temperatures (see Supplementary Note 4 for more temperatures), where  
165  $\mathbf{r}_1$  is mapped over the entire field of view and  $\mathbf{r}_2$  is fixed at position P1 (left panel) or P2 (right  
166 panel). Below  $T_c$ , the correlation functions at both positions are anisotropic with different  
167 patterns. The three-fold pattern (P1) and two-fold pattern (P2) at 22.2 K reflect the local  
168 geometry of the boundary. Above  $T_c$ , on the other hand, the patterns become largely isotropic  
169 since the inhomogeneity effect is less important when the long-range magnetic order is destroyed.  
170 Figure 3b and 3c show a line cut of the correlation function map for position P1 and P2,  
171 respectively, along the dotted lines in Fig. 3a at varying temperatures. The correlation functions  
172 feature both an oscillatory component and an exponential envelope function, and can be well  
173 fitted by  $\cos(\frac{\pi l}{\lambda}) e^{-\frac{l}{\xi}}$  (solid lines in Fig. 3b and 3c). Here  $l$  is the distance to position P1 or P2;  
174 and  $\lambda$  and  $\xi$  represent the oscillation half-period and the decay length, respectively. An enhanced  
175 correlation decay length  $\xi$  is seen for both positions near  $T_c$ , but it peaks at slightly different  
176 temperatures (Fig. 3e), again showing the effect of sample inhomogeneity. The oscillatory  
177 component in the spatial correlation is not well understood. It suggests the presence of a domain-  
178 like structure with a characteristic width  $\lambda$  (Fig. 3d). The extracted width  $\lambda$  is larger than the  
179 sample size for  $T$  below  $\sim 21$  K (see Supplementary Movie 2) and decreases rapidly when  $T$   
180 approaches  $T_c$ . Such a behavior is consistent with the prediction that in 2D ferromagnets even a  
181 tiny dipole-dipole interaction becomes increasingly important near  $T_c$  and leads to a decreased  
182 domain size<sup>28,29</sup>.

183 The sharp temperature dependence of the critical magnetization fluctuations in 2D magnets  
184 demonstrated above is fundamentally different from the behavior of non-critical Brownian  
185 fluctuations in minuscule systems<sup>30</sup> and opens up opportunities for their efficient control. We  
186 fabricate field-effect devices of monolayer CrBr<sub>3</sub> with a graphene gate and contact, and achieve  
187 tuning of  $T_c$  at a rate of  $\sim -0.4$  K/V by applying a gate voltage  $V_g$ . Gating primarily introduces  
188 doping and tunes the intralayer magnetic exchange coupling in monolayer CrBr<sub>3</sub> (See Methods  
189 and Supplementary Note 6)<sup>13</sup>. Because critical fluctuations depend sensitively on how close the  
190 system is from the critical point, they can be effectively controlled by  $V_g$  for instance at a fixed  
191 sample temperature. Figure 4a shows the magnetization dynamics of device S2 under different  
192 gate voltages at 17.90 K (below  $T_c$  at  $V_g=0$  V). The corresponding gate-dependent fluctuation  
193 amplitude and the correlation decay time are shown in Fig. 4b. At this temperature, application  
194 of  $V_g < 0.5$  V can turn on the critical fluctuations that are absent without  $V_g$ . In Fig. 4c we  
195 demonstrate switching of a magnetic bit by harnessing the critical fluctuations. In the unshaded  
196 regions ( $V_g=0$  V), the critical fluctuations are absent, and the magnetic bit stays either in state “1”  
197 or state “0” as prepared. In the yellow shaded regions ( $V_g=0.4$  V), the fluctuations are activated.  
198 Using real-time feedback control (i.e. removing  $V_g$  at the right moment according to the real-time  
199 magnetization measurement), one can set the bit into a desired state (see Supplementary Note 5  
200 for similar switching operation in sample S1 with optical control). Because the spontaneous and  
201 stochastic fluctuations are the only driving force here, the outcome cannot be deterministic  
202 without feedback. For example, Fig. 4d shows that the magnet can be stochastically switched  
203 between state “1” and “0” by applying square gate voltage pulses with an amplitude 0.4 V and a  
204 pulse width 50 ms (see Methods). Deterministic logical operations can, however, be achieved by  
205 making one measurement after each gate pulse and stop the procedure until the desired state is

206 reached; and in principle the only energy cost of the switching operation is in measuring the  
207 system's state. This critical-fluctuation-based concept can therefore potentially provide a  
208 solution to efficient magnetic processing and storage, such as logic gates and race-track memory.

209 We note that complete understanding of the critical fluctuations in 2D CrBr<sub>3</sub> will require a  
210 sophisticated model that at least accounts for finite magnetic anisotropy, dipolar interaction and  
211 the boundary effect in a framework beyond the mean-field theory. It would be interesting to see  
212 the potential crossover between different universality classes or the emergence of hidden phases  
213 near the critical point<sup>28,29</sup> and real-time imaging of fluctuations can provide invaluable  
214 information to that purpose. Nevertheless, our observation clearly demonstrates an unexplored  
215 and unique aspect of 2D layered ferromagnets compared to their bulk counterparts. Combining  
216 the opportunity to access the critical fluctuation regime in 2D, the device-compatibility of  
217 layered materials to enable electrical readout and manipulation, and the magneto-optical imaging  
218 technique to monitor magnetization fluctuations in real-time, 2D layered magnets provide an  
219 attractive platform for studying spin fluctuations and critical phenomena, as well as for  
220 fluctuation-based apparatuses such as dissipationless memories, Brownian motors and reservoir  
221 computation.

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286 **Methods:**

287 Sample preparation: Thin flakes of CrBr<sub>3</sub> encapsulated by hexagonal boron nitride (hBN) on  
288 both sides were employed in this study. The hBN-CrBr<sub>3</sub>-hBN stacks were fabricated using a dry  
289 transfer method<sup>31</sup>. CrBr<sub>3</sub> and hBN flakes were first exfoliated from their bulk crystals onto  
290 silicon substrates with a 300-nm oxide layer. The thickness of hBN and thicker flakes of  
291 CrBr<sub>3</sub> was determined by atomic force microscopy. The thickness of 2D CrBr<sub>3</sub> was  
292 determined from the calibrated optical reflection contrast (see Supplementary Note 1). A  
293 stamp made of polycarbonate (PC) on polydimethylsiloxane (PDMS) was used to pick up the  
294 top hBN flake, the CrBr<sub>3</sub> flake and the bottom hBN flake from Si substrates in sequence at  
295 40°C. For gate-tunable monolayer CrBr<sub>3</sub> device S2, two additional bilayer graphene flakes were  
296 exfoliated from bulk crystals and picked up during the assembly, serving as the contact and back  
297 gate, respectively. The complete structure was then stamped onto a clean silicon substrate with  
298 pre-patterned gold electrodes at 120°C to delaminate the PC film with the heterostructure from  
299 the PDMS layer. The residual PC was dissolved in chloroform at room temperature. The  
300 transfer process was performed inside a nitrogen gas filled glovebox with less than 1 part per  
301 million (ppm) oxygen and moisture to avoid degradation of the CrBr<sub>3</sub> flakes.

302 Real-time MCD imaging: A 409-nm laser diode (Thorlabs L405P150) was used as the probe  
303 light source. The photon energy was chosen to be slightly below the absorption edge of CrBr<sub>3</sub> to  
304 enhance the magneto-optical sensitivity. Two Glan-Taylor polarizers (Thorlabs GT10), one  
305 broadband half-wave plate (Thorlabs AHWP05M-600) and one broadband quarter-wave plate  
306 (Thorlabs AQWP05M-600) were used to control and analyze the polarization of the probe beam.  
307 The probe beam transmitted through a beamsplitter (Thorlabs BS028) and an objective (Olympus  
308 LUCPlanFLN 40x, NA = 0.6), and impinged on the sample. The beamsplitter was mounted on a

309 post by epoxy to minimize strain and polarization distortion from strain. Samples were mounted  
310 in a Montana cryostation (Standard series). An out-of-plane magnetic field was applied through a  
311 home-made coil surrounding the sample chamber with a maximum field strength of 20 Oe. The  
312 incident intensity of the probe beam on the sample was  $0.09 \mu\text{W}/\mu\text{m}^2$  for all the measurements  
313 presented in the text. For control experiment with different probe beam intensities, see  
314 Supplementary Note 8. A 2D electron-multiplying CCD camera (Princeton Instruments, ProEM  
315 512x512) was used to detect reflected light for real-time imaging.

316 Measurement of magnetic properties away from  $T_c$ : The hysteresis loop of  $\text{CrBr}_3$  monolayer (Fig.  
317 1e) and the DC magnetic susceptibility of both the  $\text{CrBr}_3$  monolayer and the thin bulk  $\text{CrBr}_3$  (Fig.  
318 1f and 1g) were obtained from the magnetization-magnetic field (M-H) dependence under an  
319 external magnetic field provided by the home-made coil. The magnetization of each flake was  
320 obtained by averaging the magneto-optical contrast over a  $3 \times 3 \mu\text{m}$  region in the center of the  
321 flake. For temperatures close to  $T_c$ , the M-H curve of monolayer  $\text{CrBr}_3$  becomes ill-defined due  
322 to the stochastic critical spin fluctuations at the macroscopic scale. Remanent magnetization in  
323 Fig. 1h is obtained by averaging the magneto-optical contrast over the  $3 \times 3 \mu\text{m}$  region and 70  
324 second to average out fluctuations.

325 Extraction of the correlation functions from real-time images: The temporal and spatial  
326 correlation functions (Fig. 2 and 3) were obtained directly from the real-time images of the  
327 magnetization fluctuations following the definition given in the text. The average was calculated  
328 as a time average over 5000 frames (taken at 70 frames per second) at each temperature. We note  
329 that, owing to the importance of the spatial inhomogeneity effect, common analysis methods that  
330 require spatial averaging, such as the Fourier transform<sup>32</sup>, become not applicable. The time  
331 average adopted in this study is closer to the original definition of the ensemble average given

332 the uniformity in time at equilibrium, and can directly provide information on the effects of  
333 spatial inhomogeneity.

334 Unusually strong critical slowing-down in 2D CrBr<sub>3</sub>: Although critical fluctuations in principle  
335 can occur at any timescale, critical dynamics observed in solid-state spin systems is usually  
336 faster than a millisecond<sup>4,33</sup>. To further verify the remarkable critical slowing-down in 2D CrBr<sub>3</sub>  
337 observed in real-time imaging, we performed AC susceptibility measurement as an independent  
338 probe. AC susceptibility is commonly used to determine the relaxation time of a system and its  
339 critical dynamics<sup>4,5</sup>. Prominent critical slowing-down is observed in monolayer CrBr<sub>3</sub> with  
340 timescales over a hundred milliseconds, but not in a ~ 10 nm thin bulk reference sample (see  
341 Supplementary Note 7). The good agreement between the AC susceptibility and real-time  
342 imaging measurements further confirms that the observed fluctuations are from the intrinsic  
343 critical behaviors of 2D CrBr<sub>3</sub>. The differences between the monolayer and bulk CrBr<sub>3</sub> samples  
344 support enhanced critical fluctuations in 2D.

345 Magnetic switching of 2D CrBr<sub>3</sub>: Measurement was performed to determine the state of a  
346 magnetic bit following the application of a square gate pulse that temporarily enables critical  
347 fluctuations. Gate pulses of variable amplitude and width were generated by a digital delay/pulse  
348 generator (Stanford Research DG535), which was triggered by a data acquisition (DAC) card  
349 (National instruments USB-6212). In Fig. 4d, the gate pulses have an amplitude 0.4 V and a  
350 pulse width 50 ms. The DAC card also triggered the electron-multiplying CCD camera to  
351 synchronize the electrical control and the MCD imaging. Measurement of the magnetization of  
352 the bit was made 200 ms after the start of each pulse to ensure that  $V_g$  and hence critical  
353 fluctuations are turned off. The above process was repeated with periodicity of 300 ms to obtain  
354 measurement sequences as exemplified in Fig. 4d.

355 **Data availability:**

356 The data that support the findings of this study are available within the paper and its  
357 Supplementary Information. Additional data are available from the corresponding authors upon  
358 request.

359 **References for Methods**

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367

368 **Competing interests:**

369 The authors declare no competing interests.

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381 **Author contributions**

382 C.J. conceived the experiment. C.J. and Z.T. developed the measurement technique and  
383 performed the experiment and analysis; Z.T. prepared the samples and K.K. fabricated the  
384 devices. K.W. and T.T. grew the bulk hBN crystals. C.J., K.F.M. and J.S. co-wrote the  
385 manuscript. All authors discussed the results and commented on the manuscript.

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387

388 **Figure captions**

389 **Figure 1 | Polarization-enhanced magnetic circular dichroism (MCD) imaging of 2D CrBr<sub>3</sub>.**

390 **a**, Illustration of the experimental setup. Blue and red beams represent illumination light from the  
391 laser and scattered light from the sample. They have different effective numerical-apertures.  
392 HWP: Half-wave plate. QWP: Quarter-wave plate. **b-d**, Optical microscopy image (**b**) and  
393 polarization-enhanced MCD image (**c**, **d**) of a monolayer CrBr<sub>3</sub> sample S1 (white dashed box).  
394 The MCD image shows giant optical contrast of  $\pm 60\%$  for the positive (**c**) and negative (**d**)  
395 remanent magnetization. Green dashed box indicates another 2D CrBr<sub>3</sub> flake nearby. Scale bar is  
396 5  $\mu\text{m}$ . **e**, Optical contrast of the monolayer against magnetic field shows a hysteresis loop at  
397 temperatures below  $T_c$  (22.3 K) and paramagnetic behavior above  $T_c$ . **f**, **g**, Temperature  
398 dependence of the DC magnetic susceptibility of the monolayer (red) and a bulk of  $\sim 10$  nm  
399 thickness (black) (**f**), and the corresponding power law fitting as described in the text (**g**). (**h**)  
400 Temperature-dependent remanent magnetization amplitude of the monolayer shows a sharp drop  
401 near  $T_c$ . Results in **e-h** are the averaged properties of a 3 x 3  $\mu\text{m}$  area near the center of the  
402 sample.

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407 **Figure 2 | Real-time imaging of critical fluctuations in 2D CrBr<sub>3</sub>.** **a**, Temperature-dependent  
408 amplitude map of magnetization fluctuations in sample S1. The critical fluctuations emerge first  
409 at the corners of the sample and exist in a narrow temperature range of  $\sim 0.5$  K around the  
410 critical point (22.3 K). **b**, Map of fluctuation correlation decay time at representative  
411 temperatures. **c, d**, Magnetization time trace (**c**) and temporal correlation function (**d**) of position  
412 P1 (green circle in **a**) at varying temperatures, showing prominent slowing-down around 21.8 K.  
413 The temporal correlation functions are normalized to 1 at zero delay. **e**, Temperature dependence  
414 of the correlation decay time (symbols) and uncertainty (error bars) from exponential function  
415 fitting of the temporal correlation function of **d**. Error in temperature is smaller than the symbol  
416 size. Vertical dashed lines represent the average  $T_c$  of the sample.

418 **Figure 3 | Spatial correlation function.** **a**, Spatial correlation function for position P1 (left) and  
419 P2 (right) at representative temperatures. The correlation patterns are anisotropic and distinct  
420 between the two points below  $T_c$ , and largely isotropic and similar above  $T_c$ . **b, c**, Normalized  
421 spatial correlation function for P1 (**b**) and P2 (**c**) along the dotted lines in **a**, which are roughly  
422 perpendicular to the domain-like structures. Symbols are experiment and solid lines are fits  
423 described in the text. Results for different temperatures are displaced vertically and successively  
424 by 0.6 for clarity. **d, e**, Domain width (**d**) and correlation decay length (**e**) obtained from fitting  
425 the spatial correlation function of P1 (black) and P2 (red). Error bars are standard deviations of  
426 the fitting parameters. Vertical dashed lines represent the average  $T_c$  of the sample. The  
427 characteristic domain width shows a monotonic decrease for both positions as temperature  
428 approaches  $T_c$  from below. The correlation decay length peaks near  $T_c$ , but deviates slightly from  
429 P1 to P2 indicating spatial inhomogeneity.

430

431 **Figure 4 | Electrical control of the critical fluctuations.** **a**, Magnetization time traces of  
432 position P3 in monolayer  $\text{CrBr}_3$  device S2 with graphene back gate. The gate voltage  $V_g$   
433 dramatically changes the critical fluctuations through tuning  $T_c$ . **b**, Gate-dependent amplitude  
434 (black) and correlation decay time (red) of critical fluctuations extracted from the time traces in **a**.  
435 **c**, At 17.90K (slightly below  $T_c$  at  $V_g = 0\text{V}$ ), the critical fluctuations are absent without gate  
436 voltage (unshaded regions) and the magnetization stays in state “1” or “0” (red dashed lines). A  
437  $V_g (=0.4\text{V})$  is used to temporarily turn on the critical fluctuations (yellow shaded regions). A  
438 specific state is written into the magnet by removing gate voltage at the right moment according  
439 to the real-time magnetization measurements. **d**, Switching between state “1” and “0” by  
440 applying square gate voltage pulses with an amplitude 0.4 V and a pulse width 50 ms at 17.90K  
441 (see text and Methods for details). A representative sequence of 100 measurements is shown.







