

pubs.acs.org/JPCC Article

# Probing Enhanced Exciton Diffusion in a Triplet-Sensitized Organic Photovoltaic Cell

Kaicheng Shi, Ian J. Curtin, Andrew T. Healy, Tao Zhang, Deepesh Rai, David A. Blank, and Russell J. Holmes\*



Cite This: J. Phys. Chem. C 2020, 124, 3489–3495



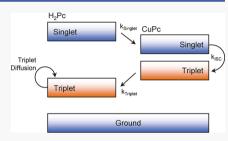
**ACCESS** I

Metrics & More

Article Recommendations

sı Supporting Information

**ABSTRACT:** We examine exciton diffusion in a triplet-sensitized organic photovoltaic cell, where transport occurs via the long-lived triplet state of a fluorescent electron donor. While the triplet state is optically dark, it is populated via sensitization by a guest species capable of intersystem crossing. Here, the host material is metal-free phthalocyanine ( $H_2Pc$ ), and the triplet-sensitizing guest is copper phthalocyanine (CuPc). Optical excitation of  $H_2Pc$  leads to the generation of singlet excitons which rapidly undergo energy transfer to CuPc. Excitons on CuPc undergo intersystem crossing to the triplet state followed by energy transfer back to the  $H_2Pc$  triplet state. The exciton diffusion length ( $L_D$ ) is extracted using an internal quantum efficiency ratio



methodology that permits accurate device-based measurements of exciton transport even in the presence of geminate recombination losses. The donor layer  $L_{\rm D}$  varies with composition with a maximum  $L_{\rm D}$  of (13.4  $\pm$  1.6 nm) observed at 20 vol % CuPc, an almost 60% increase over the case of the mobile  $H_2$ Pc singlet. Despite this increase, further improvements may be possible as the neat-film  $H_2$ Pc triplet  $L_{\rm D}$  is estimated to exceed (20.7  $\pm$  5.0) nm.

#### ■ INTRODUCTION

The design of organic photovoltaic cells (OPVs) is strongly dictated by the short exciton diffusion length  $(L_{\rm D})$  of the component active materials. Indeed, it is the short  $L_{\rm D}$  that led to the development of the bulk heterojunction architecture to increase exciton harvesting. <sup>1-10</sup> Increased active material  $L_{\rm D}$  is desirable as it could relax the spatial constraints on film morphology in bulk heterojunctions or permit the realization of efficient, planar heterojunction OPVs. Several works have previously examined the potential for increased  $L_{\rm D}$  through the use of long-lived triplet excitons. <sup>11-16</sup> The challenge with the use of such states is their low optical absorption, making them less accessible under optical pumping.

Prior work has demonstrated the ability to overcome the challenge of low optical absorption through the use of a hostguest, triplet-sensitized OPV. 16-27 In a triplet-sensitized OPV, a guest molecule capable of rapid singlet-triplet intersystem crossing is added to the donor layer in order to sensitize triplets on the fluorescent donor host. In this configuration, photogenerated host singlets undergo energy transfer to a triplet-sensitizing guest, which rapidly forms triplets that are subsequently transferred back to the long-lived host triplet state. In prior demonstrations, the guest is a heavy-metalcentered phosphor capable of rapid intersystem crossing. 16-27 For example, Luhman and Holmes used a host-guest pairing of N,N'-bis(naphthalen-1-yl)-N,N'-bis(phenyl)benzidine (NPD) and tris[2-phenylpyridinato-C2,N]iridium(III) (Ir- $(ppy)_3$ ) to demonstrate an increase in  $L_D$  from  $(6.5 \pm 0.3)$ to  $(11.8 \pm 0.6)$  nm at 5% doping. 18 Rand et al. investigated the host—guest pairing of phenyl-substituted poly(p-phenylene vinylene) (PPV) and platinum octaethylporphyrin (PtOEP) and demonstrated an increase in  $L_{\rm D}$  from (4  $\pm$  1) to (9  $\pm$  1) nm for an optimal doping of 5%. This approach has been further applied to bulk heterojunction OPVs realizing significant increases in photocurrent. Rangel and Tang incorporated PtOEP into poly[2-methoxy-5-(2-ethyl-hexyloxy)-1,4-phenylenevinylene (MEH-PPV) and found an increase in  $L_{\rm D}$  from 2.0 to 5.0 nm. Despite prior demonstrations of enhanced device performance using triplet sensitizers, there are several challenges associated with quantifying  $L_{\rm D}$  in these systems.

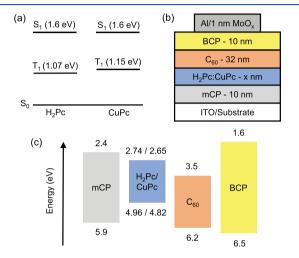
Previous demonstrations of triplet-sensitized OPVs have often relied on fitting device external quantum efficiency  $(\eta_{\rm EQE})$  spectra to extract  $L_{\rm D}$ . Device-based methods are employed as the dark host triplets are not accessible using conventional photoluminescence (PL) quenching measurements. Values of  $L_{\rm D}$  extracted from fitting  $\eta_{\rm EQE}$  spectra are frequently underestimates due to unaccounted geminate recombination losses. A second challenge with previous work is the open question of how large an increase in  $L_{\rm D}$  is practically possible. Prior reports have not separately characterized the triplet  $L_{\rm D}$  of the fluorescent host nor

Received: November 27, 2019 Revised: January 16, 2020 Published: February 3, 2020



typically is there discussion of how much exciton transport occurs via the triplet of the sensitizer. To address these issues, a device-based measurement technique for  $L_{\rm D}$  is employed that is not impacted by unknown geminate recombination losses. This approach has been previously vetted against PL-based methods for luminescent systems, showing good agreement for the extracted values of  $L_{\rm D}$ . We further characterize a lower-limit on the diffusion length of the host triplet allowing some assessment of the upper limit for  $L_{\rm D}$  in sensitized systems.

Here, exciton transport is examined for a composite donor layer consisting of a metal-free phthalocyanine ( $H_2Pc$ ) host and a copper phthalocyanine (CuPc) sensitizer as a function of the composition (Figure 1). The host  $H_2Pc$  has a singlet



**Figure 1.** (a) Singlet (S) and triplet (T) excitonic energy levels of  $H_2Pc$  and CuPc. (b) Device architecture for  $H_2Pc$ : $CuPc-C_{60}$  planar heterojunction OPVs. The  $H_2Pc$ :CuPc thickness x varies from 8 to 28 nm. (c) Energy-level diagram for the devices in (b). The LUMO levels are estimated using the optical gap for mCP and taken from prior reports of inverse photoelectron spectroscopy for  $H_2Pc$ , CuPc,  $C_{60}$ , and BCP.

energy level of  $E_s = 1.6$  eV above the ground state calculated as the cut-off wavelength of the extinction coefficient and a triplet energy level  $E_T = 1.07 \text{ eV}$ . The guest CuPc has nominally the same singlet energy as H<sub>2</sub>Pc and a slightly larger triplet energy  $E_{\rm T}$  = 1.13–1.15 eV, permitting singlet energy transfer to CuPc and subsequent triplet energy transfer to H<sub>2</sub>Pc. 36,37 Using time-resolved two-photon photoemission, Dutton and Robey have previously shown that intersystem crossing in CuPc occurs on a subpicosecond timescale.<sup>38</sup> Using transient absorption spectroscopy, Caplins et al. found that in vapordeposited thin films of CuPc, intersystem crossing occurs about 10<sup>4</sup> times faster than natural decay.<sup>39</sup> This suggests that the triplet yield in CuPc thin films is near-unity, allowing CuPc to act as a potentially efficient sensitizer to populate triplets in H<sub>2</sub>Pc. The use of CuPc as a sensitizer is advantageous in that unlike Ir- and Pt-phosphors, CuPc is highly absorbing, meaning the inclusion of the sensitizer does not reduce donor layer absorbance.

# **■ EXPERIMENTAL SECTION**

Figure 1b shows the OPV layer structures of interest for this study with a host–guest donor layer of H<sub>2</sub>Pc:CuPc and an acceptor layer of C<sub>60</sub>. The corresponding highest occupied and lowest unoccupied molecular orbital (LUMO) energy levels

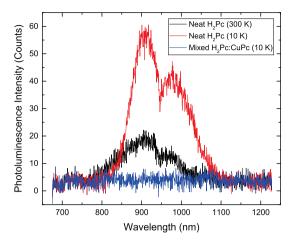
for each layer are shown in Figure 1c. $^{40-42}$  Layers of 1,3-bis(N-carbazolyl)benzene (mCP) and bathocuproine (BCP) act as exciton blocking layers to prevent quenching at the electrodes. A 1 nm-thick layer of  $MoO_x$  is deposited before the 100 nm-thick Al cathode to increase the cathode work function and reduce the built-in electric filed, minimizing exciton bulk ionization process at short circuit.

Layers were deposited by high vacuum thermal evaporation on indium-tin-oxide-coated glass substrates with a sheet resistance of  $8-12~\Omega/\Box$ . Substrates were sequentially cleaned with tergitol, acetone, and isopropanol, followed by exposure to UV—ozone ambient for 15 min prior to deposition. The circular device active area is 0.785 mm² defined by the cathode area. The materials mCP (99.5%), CuPc (99%), Ir(ppy)<sub>3</sub> (99%), and tris(4-carbazoyl-9-ylphenyl)amine (TCTA) (99.5%) were purchased from Luminescence Technology Corporation.  $C_{60}$  (99.9%) was purchased from MER Corporation.  $H_2$ Pc (98%) was purchased from Sigma-Aldrich Corporation. BCP (98%), MoO<sub>3</sub> (99%), and Al shot (99.999%) were purchased from Alfa Aesar. NPD (99%) was synthesized by the Dow Chemical Company and purified by temperature-gradient sublimation.

Thin film PL spectra were collected under pumping by a laser at a wavelength of  $\lambda = 405$  nm using a Princeton Instruments FERGIE spectrometer. Low temperature measurements were carried out in a liquid He optical cryostat. Film thickness, optical constants (except H<sub>2</sub>Pc and CuPc), and reflectivity spectra were measured using a J.A. Woollam variable-angle spectroscopic ellipsometer. Reflectivity measurements were performed at an incident angle of 15° to the substrate normal. The optical constants of neat H<sub>2</sub>Pc and CuPc were extracted by fitting transmission (at normal incidence) and reflectance (at 15°) using an optical transfer matrix model.<sup>47</sup> The optical constants of mixed H<sub>2</sub>Pc:CuPc films were calculated using a linear superposition based on neat-film optical constants. Device external quantum efficiency curves were measured under illumination from a 300 W Oriel Xe lamp equipped with a Cornerstone 130 1/8 m monochromator and a Stanford Research Systems SR540 optical chopper. Error bars for  $L_{\rm D}$  represent a 95% confidence interval extracted from fitting.

#### ■ RESULTS AND DISCUSSION

Prior to characterizing exciton diffusion in OPVs based on mixtures of H<sub>2</sub>Pc:CuPc, it is important to first verify that the triplet of the host is in fact being pumped via energy transfer from CuPc. A measurement of increased  $L_{\rm D}$  in mixtures alone is not sufficient as the lifetime and diffusivity of the H<sub>2</sub>Pc singlet are both likely a function of the composition. 48-53 To first test the singlet energy transfer from H<sub>2</sub>Pc to CuPc, PL spectra for a neat film of H<sub>2</sub>Pc (~300 nm) and an equal mixture of H<sub>2</sub>Pc:CuPc (~200 nm) were compared (Figure 2). In the neat film, H<sub>2</sub>Pc is found to exhibit a measurable PL at temperatures of T = 300 K and T = 10 K with an increased intensity as the temperature is reduced.<sup>54</sup> In contrast, no PL is detected from the mixture, suggesting that excitons generated on H<sub>2</sub>Pc are effectively transferred to the CuPc guest. Further considering the ultrafast intersystem crossing rate ( $\sim$ 500 fs) of CuPc, it is likely that all the singlets are converted into CuPc triplets. 38,39 We further argue that given the exothermic energetic alignment for triplet energy transfer from CuPc to H<sub>2</sub>Pc, a thermodynamically significant H<sub>2</sub>Pc triplet population should accumulate because of the sensitization effect.



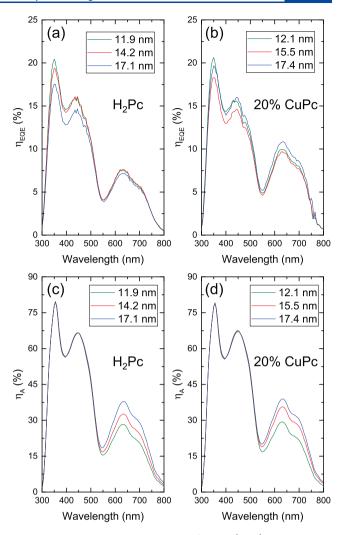
**Figure 2.** Thin film PL spectra for neat  $H_2Pc$  ( $\sim 300$  nm) and films of  $H_2Pc$  doped with 50 vol % CuPc ( $\sim 200$  nm) films on Si substrates. The pump wavelength is  $\lambda = 405$  nm.

As noted earlier, the  $L_{\rm D}$  of dark excitons is often estimated by fitting  $\eta_{\rm EQE}$  spectra. <sup>29,31,47,55</sup> Photoconversion can be divided into four sequential processes each with a corresponding efficiency namely optical absorption  $(\eta_A)$ , exciton diffusion and dissociation  $(\eta_D)$ , charge-transfer (CT) state separation  $(\eta_{\rm CS})$ , and free carrier collection  $(\eta_{\rm FC})$ . While  $\eta_{\rm FC} \approx 100\%$ under short-circuit conditions,  $\eta_{CS}$  is not rigorously unity, and charge separation competes with geminate recombination.  $^{33-35,56-58}$  As such, assuming  $\eta_{\rm CS} \approx 100\%$  and fitting  $\eta_{\mathrm{EQE}}$  only for optical generation  $(\eta_{\mathrm{A}})$  and exciton diffusion  $(\eta_{\rm D})$  can yield an underestimate to the material-relevant  $L_{\rm D}$ . Here, an alternate device-based method is applied that is based on fitting ratios of the donor-to-acceptor internal quantum efficiency  $(\eta_{\rm IQE})$  as a function of the layer thickness. The  $\eta_{\rm IQE}$ is the ratio of the number of collected charge carriers to the number of absorbed photons, which equals to  $\eta_{\rm EQE}$  divided by  $\eta_A$ . This method circumvents unknown recombination losses as losses associated with the CT state are identical for the donor and acceptor, and hence cancel in the ratio to yield material-relevant  $L_{\rm D}$  as<sup>35</sup>

$$\frac{\eta_{\rm IQE}^{\rm D}}{\eta_{\rm IQE}^{\rm A}} = \frac{\eta_{\rm D}^{\rm D} \cdot \eta_{\rm CS}}{\eta_{\rm D}^{\rm A} \cdot \eta_{\rm CS}} = \frac{\eta_{\rm D}^{\rm D}}{\eta_{\rm D}^{\rm A}} \tag{1}$$

where the superscripts D and A denote donor and acceptor efficiencies, respectively. Figure 3a,b shows  $\eta_{\rm EQE}$  spectra collected at short circuit for the device in Figure 1b with donor layers consisting of either neat H<sub>2</sub>Pc or 80 vol % H<sub>2</sub>Pc:20 vol % CuPc, respectively. Figure 3c,d shows the absorption efficiency spectra ( $\eta_{\rm A}$ ) of the active layers calculated using an optical transfer matrix formalism.<sup>47</sup> Comparing the  $\eta_{\rm EQE}$  spectra, the doped device shows an increase in the efficiency of the donor relative to the acceptor. The interfacial CT state energies ( $E_{\rm CT}$ ) are nominally ~1.06 and ~1.08 eV for H<sub>2</sub>Pc-C<sub>60</sub> and CuPc-C<sub>60</sub>, respectively.<sup>59,60</sup> Prior work has suggested that despite a potentially small energetic driving force, triplet dissociation is facilitated by an entropic driving force or the presence of an interfacial dipole that may locally impact the energetic alignment.<sup>60</sup>

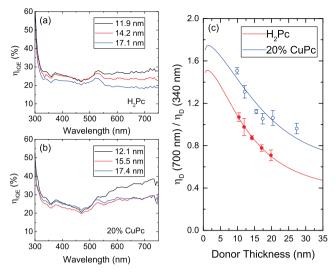
In order to calculate the change in  $L_{\rm D}$  responsible for the observed increase in donor  $\eta_{\rm EQE}$ , Figure 4a,b shows the  $\eta_{\rm IQE}$  spectra for the devices of Figure 3. The donor-to-acceptor  $\eta_{\rm IQE}$  ratios of Figure 4c are calculated by dividing  $\eta_{\rm IQE}$  at  $\lambda=700$ 



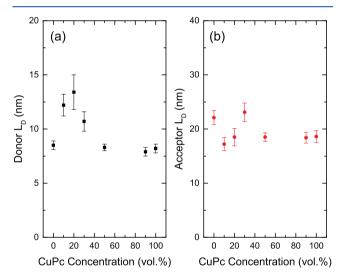
**Figure 3.** Measured external quantum efficiency  $(\eta_{\rm EQE})$  and calculated absorption efficiency  $(\eta_{\rm A})$ . The  $\eta_{\rm EQE}$  measured at short circuit as a function of the donor layer thickness for devices containing a donor layer of H<sub>2</sub>Pc (a) or 80 vol % H<sub>2</sub>Pc:20 vol % CuPc (b). (c,d) Calculated  $\eta_{\rm A}$  for the devices in (a,b), respectively.

nm (primarily donor absorption) by that at  $\lambda=340$  nm (primarily acceptor absorption). Through eq 1,  $\eta_{\rm D}$  ratios can be determined as a function of donor thickness and iteratively fit for the donor and acceptor layer  $L_{\rm D}$ . With knowledge of  $L_{\rm D}$ , the absolute  $\eta_{\rm EQE}$  can be fit using the previously unknown value of  $\eta_{\rm CS}$  as a fit parameter except for a range of  $\lambda=410-550$  nm, where prior work has noted a possible relaxation bottleneck for bulk CT state excitons in  $C_{60}$ . The associated fits of  $\eta_{\rm EQE}$  are shown in Figures S4–S10.

Using the aforementioned device-based method, donor and acceptor  $L_{\rm D}$  values are extracted as a function of the donor layer composition, as shown in Figure 5. It is worth noting that while the dissociation efficiency of the donor—acceptor CT state may vary in compositions, this would not impact the ability to extract  $L_{\rm D}$  in a self-consistent manner. The donor  $L_{\rm D}$  increases from a value of  $(8.5 \pm 0.4)$  nm for a neat film of  $H_2 Pc$  to a maximum of  $(13.4 \pm 1.6 \text{ nm})$  at 20 vol % CuPc (Figure 4c). For compositions beyond the maximum, the value of  $L_{\rm D}$  decreases with the increasing CuPc content. This trend likely suggests a trade-off between exciton transport on different states in the donor layer. If the guest concentration is small enough that the separation between  $H_2 Pc$  and CuPc



**Figure 4.** Extracting intrinsic  $L_{\rm D}$  based on fitting thickness-dependent ratios of the internal quantum efficiency. The  $\eta_{\rm IQE}$  spectra calculated as a function of the donor layer thickness from the  $\eta_{\rm EQE}$  for devices containing a donor layer of H<sub>2</sub>Pc (a) or 80 vol % H<sub>2</sub>Pc:20 vol % CuPc (b). (c) Calculated diffusion efficiency ratio ( $\lambda=700$  nm to  $\lambda=340$  nm) as a function of the donor layer thickness for the devices (a,b), respectively. The value of  $L_{\rm D}$  is extracted from the fit (solid lines).



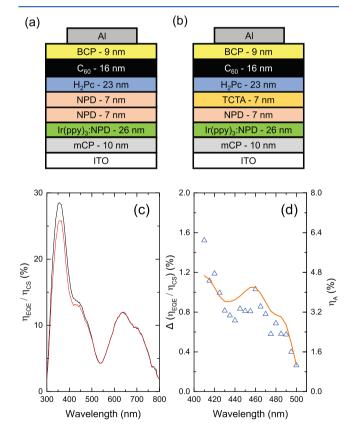
**Figure 5.** Extracted  $L_{\rm D}$  of the donor and acceptor as a function of the donor concentration. The  $L_{\rm D}$  values of the donor (a) and acceptor (b) are extracted simultaneously based on fitting thickness-dependent ratios of donor—acceptor  $\eta_{\rm IQE}$ .

molecules exceeds the singlet exciton  $L_{\rm D}$  and host–guest Förster radius, the efficiency of energy transfer to the guest is reduced and diffusion occurs via the  $\rm H_2Pc$  singlet. If instead the host ( $\rm H_2Pc$ ) concentration is small, triplets are confined to and diffuse on CuPc. It is thus reasonable to expect the optimum composition to be  $\rm H_2Pc$ -rich because the short-range Dexter energy transfer of triplets from CuPc to  $\rm H_2Pc$  is likely the rate-limiting step. The  $\rm C_{60}$  acceptor  $\rm L_D$  in Figure 5b shows no systematic variation in composition and is comparable to previously reported values. This suggests that only the diffusion efficiency of the donor layer is impacted by changes in the donor layer concentration.

It is interesting that the  $L_D$  values of neat H<sub>2</sub>Pc and CuPc are comparable despite the difference in the diffusing exciton spin

state. It was previously reported that the exciton lifetime in CuPc is about 35 times larger than that of  $\rm H_2Pc.^{39}$  This suggests that the corresponding diffusion coefficient for triplet excitons in CuPc is  $\sim 1/35$  that of singlet excitons in  $\rm H_2Pc.$  This difference may reflect the short-range Dexter energy transfer responsible for triplet diffusion. It also reemphasizes the need to populate the longer lived triplet states in a fluorescent host rather than using the shorter lived triplets present in materials with rapid intersystem crossing. Based on the extracted  $L_{\rm D}$  value of 8.2 nm and the previously reported lifetime 8.6 ns, the diffusion coefficient for triplets in CuPc is calculated to be  $7.8 \times 10^{-5}$  cm²/s, comparable to previous measurements of triplet diffusion.

An important question is whether the observed increase in the donor layer  $L_{\rm D}$  fully utilizes the long-lived triplets in  ${\rm H_2Pc}$ . Several factors must be considered when answering this question including an assessment of the intrinsic triplet  $L_{\rm D}$  in  ${\rm H_2Pc}$ . A lower limit on the triplet  $L_{\rm D}$  in a neat film of  ${\rm H_2Pc}$  is determined using a photocurrent measurement capable of selectively injecting triplets. The relevant device architectures are shown in Figure 6a,b, while the associated molecular orbital energy levels are shown in Figure S11a.  $^{36,40-42,62-64}$  In these structures, a mixed layer of NPD:Ir(ppy) $_3$  is optically pumped with photogenerated Ir(ppy) $_3$  triplets ( $E_{\rm T}=2.4$  eV) transferred to the NPD triplet state ( $E_{\rm T}=2.3$  eV). Triplets that



**Figure 6.** Extraction of the triplet diffusion length of  $H_2Pc$ . Relevant architectures enabling direct injection into the  $H_2Pc$  triplet both without (a) and with (b) a triplet blocking layer of TCTA. (c) Experimental  $\eta_{EQE}/\eta_{CS}$  spectra for devices without (black line) and with a (red line) triplet blocker. The triplet injection layer is a uniform mixture of  $Ir(ppy)_3$ :NPD. (d) Calculated  $\Delta(\eta_{EQE}/\eta_{CS})$  spectrum (symbols) and the absorption efficiency  $\eta_A$  (solid line) of the  $Ir(ppy)_3$ :NPD layer with absorption occurring only in  $Ir(ppy)_3$ .

migrate across the adjacent neat layer of NPD are subsequently injected into the triplet state of  $\rm H_2Pc$ . Injected triplets that are harvested from  $\rm H_2Pc$  will lead to an increase in external quantum efficiency over wavelengths where  $\rm Ir(ppy)_3$  absorbs. Extinction spectra for all active materials are shown in Figure S11b. A control architecture is also measured (Figure 6b) where half of the neat NPD layer is replaced with the triplet exciton blocker TCTA ( $E_{\rm T}=2.8~{\rm eV}$ ), frustrating triplet injection from NPD into  $\rm H_2Pc$ . This control structure is used to decouple the direct optical excitation of  $\rm H_2Pc$  and  $\rm C_{60}$  in Figure 6a.

To extract the triplet L<sub>D</sub> of H<sub>2</sub>Pc using the architectures of Figure 6, it is necessary to know the number of triplets injected from the neat layer of NPD into H<sub>2</sub>Pc and the number of triplets dissociated at the H<sub>2</sub>Pc-C<sub>60</sub> interface. The number of triplets injected can be calculated based on the number of photogenerated triplet excitons in Ir(ppy)<sub>3</sub> and the diffusion efficiency of the neat NPD layer. Assuming that all triplets in Ir(ppy)<sub>3</sub> are injected into the neat layer of NPD, triplet diffusion in NPD can be modeled using a previously measured value for the diffusion length ( $L_D = 25$  nm), leading to an upper limit on the injected exciton density and ultimately a lower limit on the triplet diffusion length of H<sub>2</sub>Pc.<sup>28</sup> The number of triplets dissociated from H<sub>2</sub>Pc is thus proportional to the difference in  $\eta_{\rm EOE}/\eta_{\rm CS}$  between devices with and without the triplet blocker of TCTA. Here, the  $\eta_{\rm EOE}$  is divided by  $\eta_{\rm CS}$ to decouple changes in the geminate recombination loss. The value of  $\eta_{CS}$  can be extracted by simulating the measured  $\eta_{EOE}$ spectra at the donor-absorption using the  $L_D$  values of Figure 5. The resulting ratios of  $\eta_{\rm EQE}/\eta_{\rm CS}$  are shown in Figure 6c for the architectures of Figure 6a,b. In comparing devices with and without a triplet blocking layer, a measurable increase in response is observed between wavelengths of 325 and 500 nm. The increase between 325 and 410 nm reflects absorption on TCTA, NPD, and Ir(ppy)3:NPD, leading to the injection of both singlets and triplets into H<sub>2</sub>Pc. Between 410 and 500 nm, the enhancement comes more exclusively from absorption on  $Ir(ppy)_3$  and reflects triplet injection into  $H_2Pc$ . By fitting the  $\Delta \eta_{\rm EQE}/\eta_{\rm CS}$  spectrum in Figure 6d, a lower limit for the triplet  $L_{\rm D}$  of H<sub>2</sub>Pc is obtained to be (20.7  $\pm$  5.0) nm.

The value of  $L_D$  extracted for the triplet in neat films of  $H_2Pc$ may suggest that further gains are possible with further optimization of the sensitization scheme. Since the triplet energy level difference between the host H<sub>2</sub>Pc and the guest CuPc is small ( $\sim 0.06-0.08$  eV), reverse energy transfer back to the CuPc triplet state may occur, reducing the overall  $L_{
m D}$ . Triplet confinement to H<sub>2</sub>Pc may be improved by using a sensitizer with higher triplet energy. However, it is also important to point out that the actual situation may be more complicated as the addition of CuPc to H2Pc will impact the intermolecular packing and spacing between H<sub>2</sub>Pc molecules as well as the triplet lifetime. These factors may conspire to set a different upper limit on the  $H_2Pc$  triplet  $L_D$  in mixed films. Further detailed spectroscopic studies are needed to assess these additional contributions to the maximum achievable value of  $L_{\rm D}$ .

# CONCLUSIONS

We characterize exciton transport in triplet-sensitized OPVs based on the mixed donor system of  $H_2Pc$ :CuPc as a function of composition. Using an internal quantum efficiency ratio methodology to decouple recombination losses, the composite donor layer  $L_D$  is measured as a function of the sensitizer

concentration. The measured trend indicates an exciton transport tradeoff between the energy transfer from host singlets to guest singlets and the energy transfer from guest triplets to host triplets. The optimal CuPc concentration is 20 vol % and the donor  $L_{\rm D}$  is increased from (8.5  $\pm$  0.4 nm) to (13.4  $\pm$  1.6 nm). Despite a nearly 60% enhancement, additional gains are likely possible because the triplet  $L_{\rm D}$  of H<sub>2</sub>Pc is measured to be >(20.7  $\pm$  5.0) nm. This suggest that there is still room to further exploit long-lived triplets, likely by employing a sensitizer with a higher triplet energy level to promote the formation of host triplets.

## ASSOCIATED CONTENT

# **5** Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.jpcc.9b11091.

Reflectivity of planar heterojunction OPVs, diffusion efficiency ratios, external quantum efficiency spectra, and details for triplet diffusion length measurement (PDF)

### AUTHOR INFORMATION

#### **Corresponding Author**

Russell J. Holmes — Department of Chemical Engineering and Materials Science, University of Minnesota, Minneapolis, Minnesota 55455, United States; orcid.org/0000-0001-7183-3673; Email: rholmes@umn.edu

#### **Authors**

Kaicheng Shi — Department of Chemical Engineering and Materials Science, University of Minnesota, Minneapolis, Minnesota 55455, United States

Ian J. Curtin — Department of Chemical Engineering and Materials Science, University of Minnesota, Minneapolis, Minnesota 55455, United States

Andrew T. Healy – Department of Chemistry, University of Minnesota, Minneapolis, Minnesota 55455, United States

**Tao Zhang** – Department of Chemical Engineering and Materials Science, University of Minnesota, Minneapolis, Minnesota 55455, United States

**Deepesh Rai** – Department of Chemical Engineering and Materials Science, University of Minnesota, Minneapolis, Minnesota 55455, United States

David A. Blank — Department of Chemistry, University of Minnesota, Minneapolis, Minnesota 55455, United States; orcid.org/0000-0003-2582-1537

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.jpcc.9b11091

### Notes

The authors declare no competing financial interest.

#### ACKNOWLEDGMENTS

This work was supported by the National Science Foundation (NSF) Program in Solid-State and Materials Chemistry DMR-1708177. R.J.H. would like to acknowledge support from a Leverhulme Trust Visiting Professorship at the University of Cambridge and a Visiting Fellowship at Clare Hall, University of Cambridge. The authors thank Dr. D. Wayne Blaylock at the Dow Chemical Company for synthesizing NPD.

#### REFERENCES

- (1) Peumans, P.; Uchida, S.; Forrest, S. R. Efficient bulk heterojunction photovoltaic cells using small-molecular-weight organic thin films. *Nature* **2003**, *425*, 158.
- (2) Suemori, K.; Miyata, T.; Yokoyama, M.; Hiramoto, M. Three-layered organic solar cells incorporating a nanostructure-optimized phthalocyanine: fullerene codeposited interlayer. *Appl. Phys. Lett.* **2005**, *86*, 063509.
- (3) Scharber, M. C.; Mühlbacher, D.; Koppe, M.; Denk, P.; Waldauf, C.; Heeger, A. J.; Brabec, C. J. Design rules for donors in bulk-heterojunction solar cells—Towards 10% energy-conversion efficiency. *Adv. Mater.* **2006**, *18*, 789–794.
- (4) Dennler, G.; Scharber, M. C.; Ameri, T.; Denk, P.; Forberich, K.; Waldauf, C.; Brabec, C. J. Design rules for donors in bulk-heterojunction tandem solar cells towards 15% energy-conversion efficiency. *Adv. Mater.* **2008**, *20*, 579–583.
- (5) Park, S. H.; Roy, A.; Beaupré, S.; Cho, S.; Coates, N.; Moon, J. S.; Moses, D.; Leclerc, M.; Lee, K.; Heeger, A. J. Bulk heterojunction solar cells with internal quantum efficiency approaching 100%. *Nat. Photonics* **2009**, *3*, 297.
- (6) Liang, Y.; Xu, Z.; Xia, J.; Tsai, S. T.; Wu, Y.; Li, G.; Ray, C.; Yu, L. For the bright future—bulk heterojunction polymer solar cells with power conversion efficiency of 7.4%. *Adv. Mater.* **2010**, *22*, E135—E138.
- (7) Qin, H.; Li, L.; Guo, F.; Su, S.; Peng, J.; Cao, Y.; Peng, X. Solution-processed bulk heterojunction solar cells based on a porphyrin small molecule with 7% power conversion efficiency. *Energy Environ. Sci.* **2014**, *7*, 1397–1401.
- (8) Zou, Y.; Holst, J.; Zhang, Y.; Holmes, R. J. 7.9% efficient vapor-deposited organic photovoltaic cells based on a simple bulk heterojunction. *J. Mater. Chem. A* **2014**, *2*, 12397.
- (9) Sun, D.; Meng, D.; Cai, Y.; Fan, B.; Li, Y.; Jiang, W.; Huo, L.; Sun, Y.; Wang, Z. Non-fullerene-acceptor-based bulk-heterojunction organic solar cells with efficiency over 7%. *J. Am. Chem. Soc.* **2015**, 137, 11156–11162.
- (10) Liang, N.; Sun, K.; Zheng, Z.; Yao, H.; Gao, G.; Meng, X.; Wang, Z.; Ma, W.; Hou, J. Perylene diimide trimers based bulk heterojunction organic solar cells with efficiency over 7%. *Adv. Energy Mater.* **2016**, *6*, 1600060.
- (11) Shao, Y.; Yang, Y. Efficient organic heterojunction photovoltaic cells based on triplet materials. *Adv. Mater.* **2005**, *17*, 2841–2844.
- (12) Menke, S. M.; Holmes, R. J. Exciton diffusion in organic photovoltaic cells. *Energy Environ. Sci.* **2014**, *7*, 499–512.
- (13) Mikhnenko, O. V.; Blom, P. W. M.; Nguyen, T.-Q. Exciton diffusion in organic semiconductors. *Energy Environ. Sci.* **2015**, *8*, 1867–1888.
- (14) Hedley, G. J.; Ruseckas, A.; Samuel, I. D. W. Light harvesting for organic photovoltaics. *Chem. Rev.* **2016**, *117*, 796–837.
- (15) Li, X.; Tang, M. L. Triplet transport in thin films: Fundamentals and applications. *Chem. Commun.* **2017**, *53*, 4429–4440.
- (16) Luppi, B. T.; Majak, D.; Gupta, M.; Rivard, E.; Shankar, K. Triplet excitons: improving exciton diffusion length for enhanced organic photovoltaics. *J. Mater. Chem. A* **2019**, *7*, 2445–2463.
- (17) Schueppel, R.; Uhrich, C.; Pfeiffer, M.; Leo, K.; Brier, E.; Reinold, E.; Baeuerle, P. Enhanced photogeneration of triplet excitons in an oligothiophene–fullerene blend. *ChemPhysChem* **2007**, 8, 1497–1503.
- (18) Luhman, W. A.; Holmes, R. J. Enhanced exciton diffusion in an organic photovoltaic cell by energy transfer using a phosphorescent sensitizer. *Appl. Phys. Lett.* **2009**, *94*, 153304.
- (19) Rand, B. P.; Schols, S.; Cheyns, D.; Gommans, H.; Girotto, C.; Genoe, J.; Heremans, P.; Poortmans, J. Organic solar cells with sensitized phosphorescent absorbing layers. *Org. Electron.* **2009**, *10*, 1015–1019.
- (20) Rand, B. P.; Girotto, C.; Mityashin, A.; Hadipour, A.; Genoe, J.; Heremans, P. Photocurrent enhancement in polymer: fullerene bulk heterojunction solar cells doped with a phosphorescent molecule. *Appl. Phys. Lett.* **2009**, *95*, 173304.

- (21) Lee, C.-L.; Hwang, I.-W.; Byeon, C. C.; Kim, B. H.; Greenham, N. C. Triplet exciton and polaron dynamics in phosphorescent dye blended polymer photovoltaic devices. *Adv. Funct. Mater.* **2010**, *20*, 2945–2950.
- (22) Roberts, S. T.; Schlenker, C. W.; Barlier, V.; McAnally, R. E.; Zhang, Y.; Mastron, J. N.; Thompson, M. E.; Bradforth, S. E. Observation of triplet exciton formation in a platinum-sensitized organic photovoltaic device. *J. Phys. Chem. Lett.* **2010**, *2*, 48–54.
- (23) Yang, D.; Li, W.; Chu, B.; Su, Z.; Wang, J.; Zhang, G.; Zhang, F. Enhancement of photovoltaic efficiency of phosphor doped organic solar cell by energy and electron transfer from the phosphor to C60 acceptor. *Appl. Phys. Lett.* **2011**, *99*, 193301.
- (24) Xiong, K.; Hou, L.; Wang, P.; Xia, Y.; Chen, D.; Xiao, B. Phosphor-doping enhanced efficiency in bilayer organic solar cells due to longer exciton diffusion length. *J. Lumin.* **2014**, *151*, 193–196.
- (25) Andernach, R.; Utzat, H.; Dimitrov, S. D.; McCulloch, I.; Heeney, M.; Durrant, J. R.; Bronstein, H. Synthesis and exciton dynamics of triplet sensitized conjugated polymers. *J. Am. Chem. Soc.* **2015**, *137*, 10383–10390.
- (26) Angel, F. A.; Tang, C. W. Understanding the effect of triplet sensitizers in organic photovoltaic devices. *Org. Electron.* **2016**, *30*, 247–252.
- (27) Du, L.; Xiong, W.; Cheng, S.-C.; Shi, H.; Chan, W. K.; Phillips, D. L. Direct observation of an efficient triplet exciton diffusion process in a platinum-containing conjugated polymer. *J. Phys. Chem. Lett.* **2017**, *8*, 2475–2479.
- (28) Rai, D.; Holmes, R. J. Measurement of the triplet exciton diffusion length in organic semiconductors. *J. Mater. Chem. C* **2019**, *7*, 5695–5701.
- (29) Luhman, W. A.; Holmes, R. J. Investigation of energy transfer in organic photovoltaic cells and impact on exciton diffusion length measurements. *Adv. Funct. Mater.* **2011**, 21, 764–771.
- (30) Lunt, R. R.; Giebink, N. C.; Belak, A. A.; Benziger, J. B.; Forrest, S. R. Exciton diffusion lengths of organic semiconductor thin films measured by spectrally resolved photoluminescence quenching. *J. Appl. Phys.* **2009**, *105*, 053711.
- (31) Scully, S. R.; McGehee, M. D. Effects of optical interference and energy transfer on exciton diffusion length measurements in organic semiconductors. *J. Appl. Phys.* **2006**, *100*, 034907.
- (32) Curtin, I. J.; Holmes, R. J. Decoupling photocurrent loss mechanisms in photovoltaic cells using complementary measurements of exciton diffusion. *Adv. Energy Mater.* **2018**, *8*, 1702339.
- (33) Zhang, T.; Holmes, R. J. Photovoltage as a quantitative probe of carrier generation and recombination in organic photovoltaic cells. *J. Mater. Chem. C* **2017**, *5*, 11885–11891.
- (34) Mullenbach, T. K.; Curtin, I. J.; Zhang, T.; Holmes, R. J. Probing dark exciton diffusion using photovoltage. *Nat. Commun.* **2017**, *8*, 14215.
- (35) Zhang, T.; Dement, D. B.; Ferry, V. E.; Holmes, R. J. Intrinsic measurements of exciton transport in photovoltaic cells. *Nat. Commun.* **2019**, *10*, 1156.
- (36) Lüth, H.; Roll, U.; Ewert, S. Electronic transitions in some phthalocyanine and NH-rhodanine-merocyanine films studied by inelastic-electron-tunneling spectroscopy. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1978**, *18*, 4241.
- (37) Rosenow, T. C.; Walzer, K.; Leo, K. Near-infrared organic light emitting diodes based on heavy metal phthalocyanines. *J. Appl. Phys.* **2008**, *103*, 043105.
- (38) Dutton, G. J.; Robey, S. W. Exciton dynamics at CuPc/C60 interfaces: Energy dependence of exciton dissociation. *J. Phys. Chem.* C **2012**, *116*, 19173–19181.
- (39) Caplins, B. W.; Mullenbach, T. K.; Holmes, R. J.; Blank, D. A. Femtosecond to nanosecond excited state dynamics of vapor deposited copper phthalocyanine thin films. *Phys. Chem. Chem. Phys.* **2016**, *18*, 11454–11459.
- (40) Kanno, H.; Holmes, R. J.; Sun, Y.; Kena-Cohen, S.; Forrest, S. R. White stacked electrophosphorescent organic light-emitting devices employing MoO3 as a charge-generation layer. *Adv. Mater.* **2006**, *18*, 339–342.

- (41) Zahn, D. R. T.; Gavrila, G. N.; Gorgoi, M. The transport gap of organic semiconductors studied using the combination of direct and inverse photoemission. *Chem. Phys.* **2006**, 325, 99–112.
- (42) Djurovich, P. I.; Mayo, E. I.; Forrest, S. R.; Thompson, M. E. Measurement of the lowest unoccupied molecular orbital energies of molecular organic semiconductors. *Org. Electron.* **2009**, *10*, 515–520.
- (43) Yi, Y.; Jeon, P. E.; Lee, H.; Han, K.; Kim, H. S.; Jeong, K.; Cho, S. W. The interface state assisted charge transport at the MoO3/metal interface. *J. Chem. Phys.* **2009**, *130*, 094704.
- (44) Zou, Y.; Holmes, R. J. Influence of a MoOx interlayer on the open-circuit voltage in organic photovoltaic cells. *Appl. Phys. Lett.* **2013**, *103*, 053302.
- (45) Morgan, N. T.; Zhang, Y.; Grandbois, M. L.; Bell, B. M.; Holmes, R. J.; Cussler, E. L. Mechanism for the separation of organic semiconductors via thermal gradient sublimation. *Org. Electron.* **2015**, 24, 212–218.
- (46) Morgan, N. T.; Zhang, Y.; Molitor, E. J.; Bell, B. M.; Holmes, R. J.; Cussler, E. L. Understanding rate-limiting processes for the sublimation of small molecule organic semiconductors. *AIChE J.* **2014**, *60*, 1347–1354.
- (47) Pettersson, L. A. A.; Roman, L. S.; Inganäs, O. Modeling photocurrent action spectra of photovoltaic devices based on organic thin films. *J. Appl. Phys.* **1999**, *86*, 487–496.
- (48) Kawamura, Y.; Goushi, K.; Brooks, J.; Brown, J. J.; Sasabe, H.; Adachi, C. 100% phosphorescence quantum efficiency of Ir (III) complexes in organic semiconductor films. *Appl. Phys. Lett.* **2005**, *86*, 071104
- (49) Menke, S. M.; Luhman, W. A.; Holmes, R. J. Tailored exciton diffusion in organic photovoltaic cells for enhanced power conversion efficiency. *Nat. Mater.* **2013**, *12*, 152.
- (50) Al-Kaysi, R. O.; Sang Ahn, T.; Müller, A. M.; Bardeen, C. J. The photophysical properties of chromophores at high (100 mM and above) concentrations in polymers and as neat solids. *Phys. Chem. Chem. Phys.* **2006**, *8*, 3453–3459.
- (51) Schlosser, M.; Lochbrunner, S. Exciton migration by ultrafast Förster transfer in highly doped matrixes. *J. Phys. Chem. B* **2006**, *110*, 6001–6009.
- (52) Vu, T. T.; Dvorko, M.; Schmidt, E. Y.; Audibert, J.-F.; Retailleau, P.; Trofimov, B. A.; Pansu, R. B.; Clavier, G.; Méallet-Renault, R. Understanding the spectroscopic properties and aggregation process of a new emitting boron dipyrromethene (BODIPY). J. Phys. Chem. C 2013, 117, 5373–5385.
- (53) Menke, S. M.; Holmes, R. J. Energy-cascade organic photovoltaic devices incorporating a host—guest architecture. ACS Appl. Mater. Interfaces 2015, 7, 2912–2918.
- (54) Yoshino, K.; Hikida, M.; Tatsuno, K.; Kaneto, K.; Inuishi, Y. Emission spectra of phthalocyanine crystals. *J. Phys. Soc. Jpn.* **1973**, *34*, 441–445.
- (55) Siegmund, B.; Sajjad, M. T.; Widmer, J.; Ray, D.; Koerner, C.; Riede, M.; Leo, K.; Samuel, I. D. W.; Vandewal, K. Exciton diffusion length and charge extraction yield in organic bilayer solar cells. *Adv. Mater.* **2017**, *29*, 1604424.
- (56) Baran, D.; Gasparini, N.; Wadsworth, A.; Tan, C. H.; Wehbe, N.; Song, X.; Hamid, Z.; Zhang, W.; Neophytou, M.; Kirchartz, T.; Brabec, C. J.; Durrant, J. R.; McCulloch, I. Robust nonfullerene solar cells approaching unity external quantum efficiency enabled by suppression of geminate recombination. *Nat. Commun.* **2018**, *9*, 2059.
- (\$7) Dibb, G. F. A.; Jamieson, F. C.; Maurano, A.; Nelson, J.; Durrant, J. R. Limits on the fill factor in organic photovoltaics: distinguishing nongeminate and geminate recombination mechanisms. *J. Phys. Chem. Lett.* **2013**, *4*, 803–808.
- (58) Credgington, D.; Jamieson, F. C.; Walker, B.; Nguyen, T.-Q.; Durrant, J. R. Quantification of geminate and non-geminate recombination losses within a solution-processed small-molecule bulk heterojunction solar cell. *Adv. Mater.* **2012**, *24*, 2135–2141.
- (59) Zou, Y.; Holmes, R. J. Correlation between the open-circuit voltage and charge transfer state energy in organic photovoltaic cells. *ACS Appl. Mater. Interfaces* **2015**, *7*, 18306–18311.

- (60) Piersimoni, F.; Cheyns, D.; Vandewal, K.; Manca, J. V.; Rand, B. P. Excitation of charge transfer states and low-driving force triplet exciton dissociation at planar donor/acceptor interfaces. *J. Phys. Chem. Lett.* **2012**, *3*, 2064–2068.
- (61) Rai, D.; Holmes, R. J. Investigation of excitonic gates in organic semiconductor thin films. *Phys. Rev. Appl.* **2019**, *11*, 014048.
- (62) Srivastava, R.; Joshi, L. R. The effect of substituted 1, 2, 4-triazole moiety on the emission, phosphorescent properties of the blue emitting heteroleptic iridium (III) complexes and the OLED performance: a theoretical study. *Phys. Chem. Chem. Phys.* **2014**, *16*, 17284–17294.
- (63) Baldo, M. A.; Forrest, S. R. Transient analysis of organic electrophosphorescence: I. Transient analysis of triplet energy transfer. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2000**, 62, 10958.
- (64) Nakanotani, H.; Masui, K.; Nishide, J.; Shibata, T.; Adachi, C. Promising operational stability of high-efficiency organic light-emitting diodes based on thermally activated delayed fluorescence. *Sci. Rep.* **2013**, *3*, 2127.