

Lattice dynamics and phase stability of rhombohedral antimony under high pressure

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The high pressure lattice dynamics of rhombohedral antimony have been studied by a combination of diffuse scattering and inelastic x-ray scattering. The evolution of the phonon behavior as a function of pressure was analyzed by means of two theoretical approaches: density functional perturbation theory and symmetry-based phenomenological phase transition analysis. This paper focuses on the first structural phase transition, SbI-SbIV, and the role of vibrations in leading the transition. The phonon dispersion exhibits complex behavior as one approaches the structural transition, with the branches, corresponding to the two transitions happening at high pressure in the Va elements (A7-to-BCC and A7-to-PC) both showing softening.

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I. INTRODUCTION

Antimony, like the other group Va elements arsenic, bismuth, and pressurized phosphorus, crystallizes under ambient conditions in an A7 rhombohedral structure. Under pressure it undergoes a sequence of structural phase transitions, adopting high pressure structures, which in many cases are common to those of other group Va elements (for a review see Ref. [1]), some of which have a surprising complexity as elemental solid structures. At the highest pressures, all Va elements, excluding molecular nitrogen, adopt a BCC structure. While the sequential details vary within the group, what remains unclear in all cases is if the lattice dynamics play a significant role in driving the transitions through these complex structures.

More specifically, in the case of elemental antimony the ambient pressure A7 structure ($R\bar{3}m$) known as SbI, emerges from a simple primitive cubic (PC) structure subjected to two displacements, shown in Fig. 1(a): firstly, an internal displacement of the two inter-penetrating cubic lattices along a common axis and secondly, a stretching on the 3-fold axis. The above mechanism lowers the energy of the system while doubling the unit cell size and opening an energy gap in the electronic density of states, hence the rhombohedral distortion of SbI is often referred to as Peierls distortion.

Early x-ray diffraction experiments showed that the Peierls distortion decreases at high pressure [2], in agreement with *ab initio* total energy calculations [3,4]. However, the simple cubic structure, even though continuously approached under pressure, is never stabilized in Sb. Instead, at ~ 10 GPa antimony adopts a host-guest structure (SbII) where both host

and guest have body centered tetragonal lattices (BCT) but the latter is incommensurate with the former along the tetragonal c axis [5,6]. At higher pressures (~ 28 GPa), a first-order phase transition takes place towards a BCC structure (SbIII), which is stable up to at least 43 GPa [7]. The phase diagram of antimony is further enriched by a second incommensurate monoclinic host-guest structure, known as SbIV, which appears in the narrow pressure range from 8 to 9 GPa [8].

Clear fingerprints of the above transitions have been identified in the high pressure Raman spectra of antimony. Continuous softening of the zone-center optical phonons was observed over the entire stability range of SbI up to the first transition towards the SbIV phase [9–11]. This effect has been observed in other A7 Va elements, such as As and Bi [10,12], and has been linked to the pressure-induced reduction of the Peierls-like distortion [3,11]. In addition to this pronounced softening, a large broadening of the Raman peaks was found under pressure, in agreement with *ab initio* calculations [13]. Two potential contributors have been suggested: anharmonic decay channels and a pressure-induced changes in the electron-phonon coupling.

While these investigations have demonstrated the sensitivity of the zone center phonons to both structural and electronic changes under pressure, the role played by lattice excitations in driving the phase transformations remains unclear. In particular, the phononic contribution in the Peierls distortion reduction and in the ultimate stabilization of the BCC structure needs to be clarified. Momentum-resolved phonon measurements focused on the A7 to PC and A7 to BCC phase transitions are needed to resolve this issue. Phonon dispersion curves along certain high symmetry directions have been obtained via earlier inelastic neutron scattering experiments at ambient conditions [14,15]. Nevertheless, up to date

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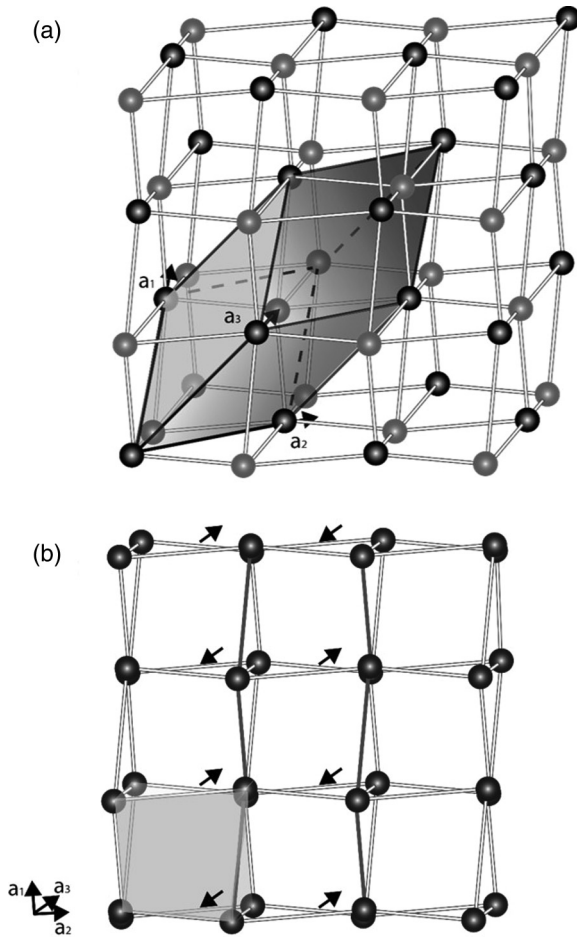


FIG. 1. (a) The A7 structure, where the gray and black balls represent the two interpenetrating FCC (face centered cubic) lattices and the shaded region corresponds to the rhombohedral unit cell. (b) A chains displacement is present in the A7 structure due to its intrinsic distortions. The black arrows show the direction of the displacement with respect to the PC structure. The shaded region corresponds to the pseudocubic cell.

a pressure-dependent, momentum-resolved lattice dynamical study has not been reported. Here we address this open issue through a comprehensive study, combining x-ray diffuse scattering (DS) and high pressure inelastic x-ray scattering (IXS), in conjunction with *ab initio* calculations and a phenomenological analysis of the displacive transformation.

II. EXPERIMENTAL METHODS

Commercially available, high quality, single crystals of antimony were obtained from SurfaceNet GMBH [16]. A platelet of $\sim 50 \mu\text{m}$ thickness was used for ambient pressure measurements while cylinders of $\sim 150 \mu\text{m}$ diameter and $\sim 50 \mu\text{m}$ thickness were laser cut for the high pressure study. In both cases the samples were etched using a HCl/HNO_3 mixture to remove surface damages. Rocking curves of $\Delta\theta \sim 0.1^\circ$ confirmed the high crystalline quality of the samples.

High pressures were obtained using a gasketed diamond anvil cell (DAC). All measurements were performed at room temperature with compressed helium used as a pressure transmitting medium and the pressure calibration was performed

using the ruby luminescence [17]. The samples were oriented with the rhombohedral [111] axis (hexagonal c axis) perpendicular to the diamond facets. Both crystallographic settings will be used to describe the SbI structure, denoted as HKL_R and HKL_H for the rhombohedral and hexagonal systems, respectively.

The DS and IXS experiments were conducted at beamlines ID23 and ID28 of the European Synchrotron Radiation Facility (ESRF), respectively. A monochromatic beam of $\lambda = 0.689 \text{ \AA}$, corresponding to 17.994 keV of energy, was used for all the DS measurements. The sample was rotated through 360° orthogonal to the incoming beam and diffuse scattering frames were collected with an angular slicing of 0.1° . A Pilatus 6M (Dectris) detector with a pixel size of $172 \times 172 \mu\text{m}^2$ was used in single photon counting mode [18]. The CrysAlis software package was used to obtain the orientation matrix and to perform a preliminary data evaluation [19]. All IXS measurements were performed using a 17.794 keV beam (with $\lambda = 0.06968 \text{ \AA}$), corresponding to an energy resolution of 3 meV, with a $15 \times 13 \mu\text{m}^2$ spot size on the sample surface. Further details about the experimental setup can be found in Ref. [20].

Additional experiments on a single crystal bismuth sample (SurfaceNet GMBH [16]) were performed using Pilatus 300K with CdTe sensor material at ID15 ESRF beamline, employing $0.178 \text{ \AA}$ wavelength and 0.1° angular slicing.

III. *AB INITIO* CALCULATIONS

Density functional theory *ab initio* calculations of the lattice dynamics were performed in order to complement and interpret the experiments using the ABINIT software package [23–26]. The antimony electronic structure in the A7 phase was calculated using the Fritz-Haber pseudopotentials by following the description in Ref. [27] with $5s^2 5p^3$ valence electrons. A grid of $8 \times 8 \times 8$ $\mathbf{k}$ points was used to integrate quantities within the Brillouin zone and obtain any electronic property. An energy cutoff of 50 Ha was employed to guarantee energy convergence to less than 10^{-5} eV/atom and stresses less than 10^{-3} GPa. The local density approximation was applied to the exchange and correlation energy [28] and the preconditioning method discussed in Ref. [29] was used to improve the density convergence. The crystal cell optimization and the interatomic force minimization were performed by following the conjugate gradient algorithm as described in Ref. [30]. The atomic coordinates, lattice vectors, and lattice parameters were fully relaxed under the constraint of constant pressures between 0 and 7 GPa. Additional calculations were performed at a constant volume of $53.5 \text{ \AA}^3$, in order to compare the quality of the calculations with Ref. [13].

The optimized cell parameters for the A7 phase are $a = 4.29 \text{ \AA}$ and $c = 11.041 \text{ \AA}$ with $z = 0.235$. A comparison with ONCVSP pseudopotentials [31,32] was performed with $5s^2 5p^3$ and $4d^{10} 5s^2 5p^3$. Differences in the optimized cell parameters are less than 0.5% and the largest difference between the vibrational spectra is less 5 cm^{-1} for the optical Γ phonon modes.

The calculated diffuse scattering maps are given by direct calculation in one-phonon scattering maps where the harmonic approximation is considered. Phonon dispersion relations along

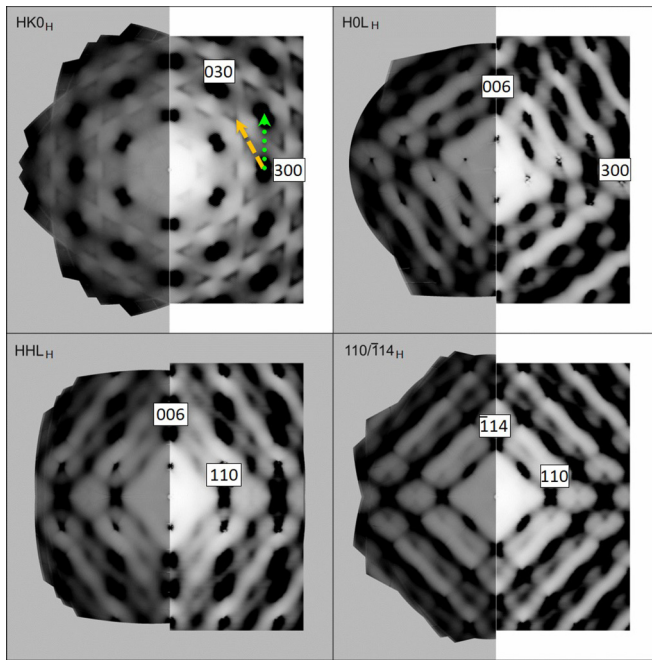


FIG. 2. Diffuse scattering maps of antimony at ambient conditions: the calculated map (right) and the experimental reconstructed reciprocal space planes (left). The yellow (dashed line) and green (dotted line) arrows correspond to two of the diffuse features: the v-shape and diffuse planes, respectively. The hexagonal setting is chosen.

selected high symmetry directions were obtained by interpolation of the dynamical matrices corresponding to a $\mathbf{q}$ -mesh grid of $8 \times 8 \times 8$ at each pressure point. These matrices were calculated within the framework of density functional perturbation theory and the linear response method [33,34].

IV. RESULTS AND DISCUSSION

a. Ambient pressure

Figure 2 shows the experimental diffuse scattering patterns of antimony, at ambient conditions as reconstructed reciprocal space planes compared with the calculated maps. From the plethora of complex diffuse patterns present in the maps of Fig. 2, we focus on those most significant to the approaching phase transition. The first notable feature is the diffuse planes in the $\langle 110/\bar{1}14 \rangle_H$ map. These planes are normal to the $\langle 110 \rangle_R$ direction or the $\langle 100 \rangle_{PC}$ direction of the pseudocubic structure. In other words, this is the direction of the chains displacement which is present in the A7 structure as an intrinsic distortion with respect to the PC structure, see the illustration by black arrows in Fig. 1(b). A 4-fold axis symmetry reminiscent of the undistorted PC structure is evident in these diffuse structures. Other notable features are in the $HK0_H$ plane: (1) lines in the $\langle 120 \rangle_H$ direction from the diffuse planes that intersect the $HK0_H$ (see the evolution of this feature in the next section dedicated to the pressure); (2) v shapes in the $\langle \bar{1}10 \rangle_H$ direction, shown in Fig. 2 by the green and yellow arrows, respectively.

The remarkable agreement between the experimental and calculated maps adds considerable credence to the accuracy of the calculations. However, the experimental intensities on

the maps are not resolved in energy, showing just a particular sensitivity for low energy features. Therefore, in order to confirm the reliability of the *ab initio* calculations and the nature of the diffuse features, IXS measurements at ambient conditions were performed. Figure 3 summarizes the ambient pressure IXS dispersion relations which show a discrepancy of less than $\pm 5\%$ when compared to the theory results. These measurements at ambient conditions complete the earlier INS studies made by Sharp and Sosnowski, which reported the phonon dispersion curves of the high-symmetry directions [14,15]. Focusing on the phonon dispersions, most directions show a low energy phonon near the zone boundary. One example is in the $[\bar{1}2\bar{1}]_R$ direction that corresponds to the v-shape diffuse features. In order to further investigate the nature of such diffuse components which have a dynamic nature, we turned to pressure dependent IXS measurements.

b. High pressure

The calculated maps of the $HK0_H$ plane at pressures up to 4.7 GPa are shown in Fig. 4. The diffuse planes visible in the $\langle \bar{1}20 \rangle_H$ directions are increasing in intensity under pressure. On the contrary, the v-shaped patterns appear not to show any pressure dependence. Figure 5 summarizes the high pressure IXS results along these two directions.

The phonon dispersion curves in the direction $[\bar{1}2\bar{1}]_R$ ($[\bar{1}10]_H$, yellow arrow in Figs. 2 and 4) display the general trend of phonon hardening with increasing pressure, see Fig. 5(b). The IXS scans at $\xi = 0.25$ at 1 GPa and 6.5 GPa provide an example of this behavior, Fig. 5(a). The optical phonon also shows hardening across the BZ, an effect which is enhanced at the zone boundary. However a softening, or a “relative softening,” meaning an absence of hardening, is observed for the acoustic branch near the BZ boundary. This is shown in detail in the scans recorded under 1 and 6.5 GPa at $\xi = 0.4$ where the acoustic phonon is unchanged in energy (~ 6 meV), see vertical lines in Fig. 5(a) on the right.

In the $[\bar{1}10]_R$ ($[\bar{1}20]_H$, green arrow in Figs. 2 and 4) direction, the LA (longitudinal acoustic) and TA (transverse acoustic) phonons behave differently under pressure, Figs. 5(c) and 5(d). There should be an anticrossing between LA and TA around $\xi = 0.43$ according to the eigenvectors provided by DFT calculations, as the phonons at lowest energy share the same polarization [22,35]. In our case, the intensity of the longitudinal phonon almost disappears while the transverse one remains measurable. The LA softens by $\sim 7\%$, whereas the TA behaves similarly to the phonon in the $[\bar{1}2\bar{1}]_R$ direction: From Γ to $\xi = 0.3$ there is a pressure-induced hardening and post $\xi = 0.4$ until the zone boundary a softening of $\sim 15\%$, Fig. 5(d).

While the calculations are generally in good agreement with the experimental results, two main differences are found. Firstly, the calculated optical phonons in the $[\bar{1}2\bar{1}]_R$ direction indicate a greater pressure induced hardening and a shift to lower energies than what was observed experimentally. Secondly, the calculations predict a greater softening of the TA than the one observed in the IXS experiment ($\sim 20\%$ compared to 15%).

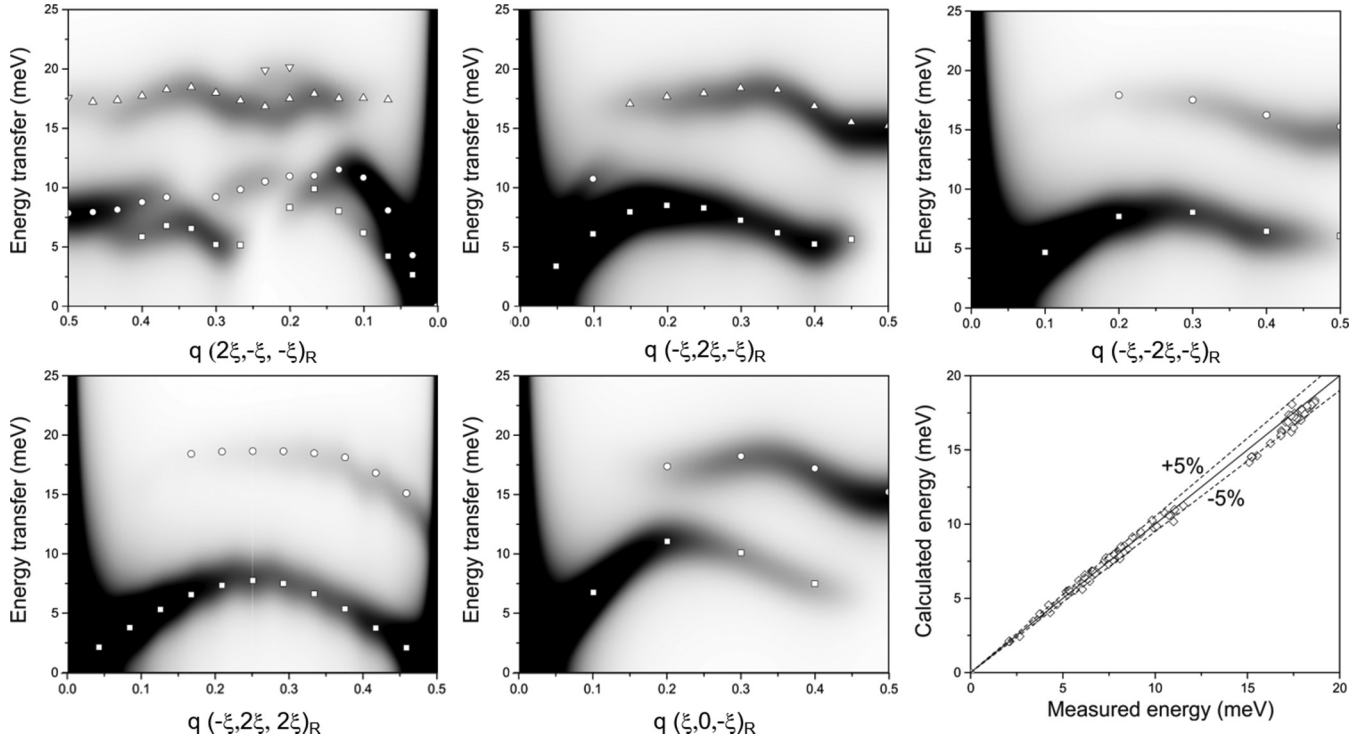


FIG. 3. Phonon dispersions in different directions of the reciprocal space at ambient conditions. Empty symbols correspond to the fitted frequencies; number of observable branches from the calculation is taken into account. Grayscale map corresponds to the theoretical dynamical structure factor $S(\mathbf{Q}, E)$, see Refs. [21,22], convoluted to the realistic resolution function. The deviation between the calculated and measured phonon energy shows a good agreement in the order of a $\pm 5\%$ error.

c. Phase transitions mechanisms and reciprocal space critical vectors

In order to understand and classify the dynamical behavior at high pressure, we analyze the corresponding phase transition mechanisms using and completing existing data [1]. To this purpose we integrate all the structural data in a unified scheme of a single parent structure, not in *direct* but in *reciprocal* space. As already mentioned in the introduction, the logical choice of a parent structure for the group Va elements is the high-pressure BCC [SbIII, space group $\text{Im}\bar{3}\text{m}(Z_{\text{III}}^P = 1)$, A2 structure type], as it is present in the P-T phase diagrams of all elements belonging to the group. Another structure common

to all Va group elements is the low pressure rhombohedral $\text{R}\bar{3}\text{m}(Z_I^P = 2)$ (SbI, A7) phase. However, SbI has been shown to be a slightly distorted version of the simple cubic structure [(pseudo)SbI, $\text{Pm}\bar{3}\text{m}(Z_{\text{PI}} = 1)$] stabilized at higher pressure in some of the group Va elements (for instance, PVI or AsII) [36,37]. Since SbI under pressure relaxes toward PC and this last structure is reached by other Va group elements, such as arsenic and phosphorus, the phase transition between the two has been included in the discussion. The corresponding critical vector for the A7 to PC transition, $\mathbf{k}_{13} = 1/2(\mathbf{b}_1 + \mathbf{b}_2 + \mathbf{b}_3)_P$, ends at the R point of the PC Brillouin zone (BZ), see Fig. 6. In order to simplify the discussion, the

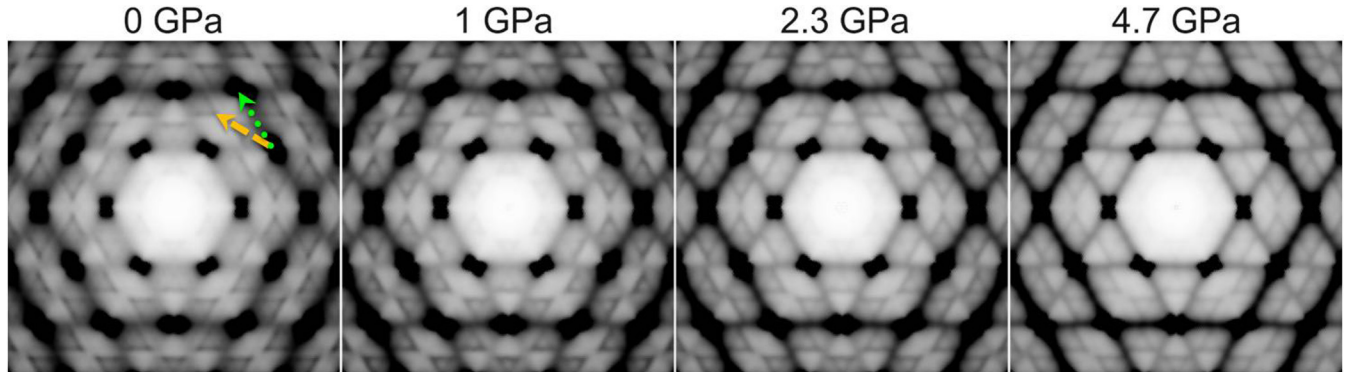


FIG. 4. $\text{HK}0_H$ or $\langle 110/112 \rangle_R$ maps reconstructed from *ab initio* calculations. Applying pressure, diffuse features in $\langle 110 \rangle_R$ directions become stronger. The dashed and dotted line (yellow and green in the online version) arrows correspond to the two directions analyzed by IXS: $[112]_R$ and $[110]_R$, respectively.

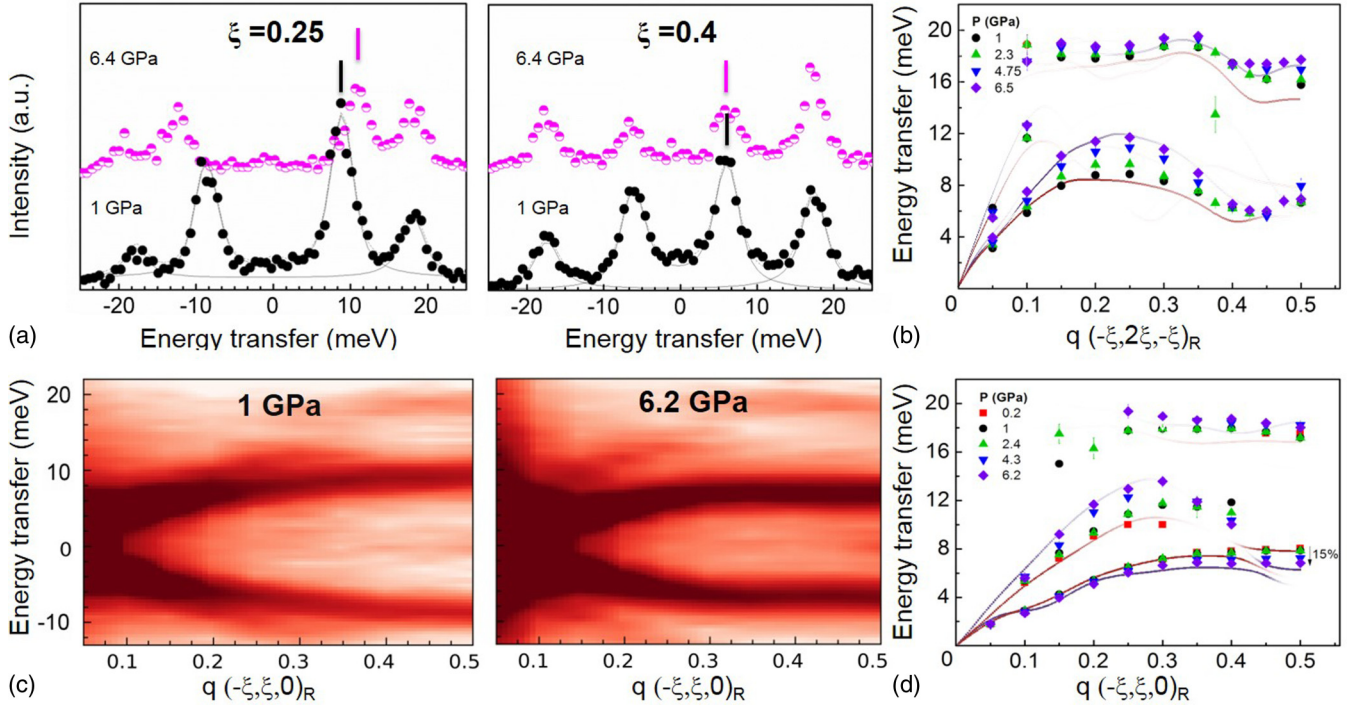


FIG. 5. (a) From left to right: IXS scans at $\xi = 0.25$, on the left, and $\xi = 0.4$, on the right, for pressures 1 GPa and 6.5 GPa, in which the lines on the top of the peaks help to visualize the phonon energy obtained from the fitting. (b) The fitting results derived from the IXS scans show the general phonon dispersion approaching the first transition. Each symbol corresponds to a specific experimental pressure. The results of the calculations are shown as lines for ambient pressure and 6.5 GPa. The line intensity is proportional to the scattering factor, which was considered for the specific direction. (c) Experimental phonon dispersions in the direction $[\bar{1}10]_R$ at different applied pressures: 1 GPa, on the left, and 6.2 GPa, on the right; (d) the fitting results derived from IXS scans at each applied pressure from 0.2 GPa to 6.2 GPa, compared to the calculations.

pseudocubic structure will substitute the distorted A7 for the following part. Thus, without loss of generality, one may consider the quartet of phases “(pseudo)SbI-SbIV-SbII-SbIII” as representative not only for Sb but for other Va group elements. Such a scheme becomes even better justified if

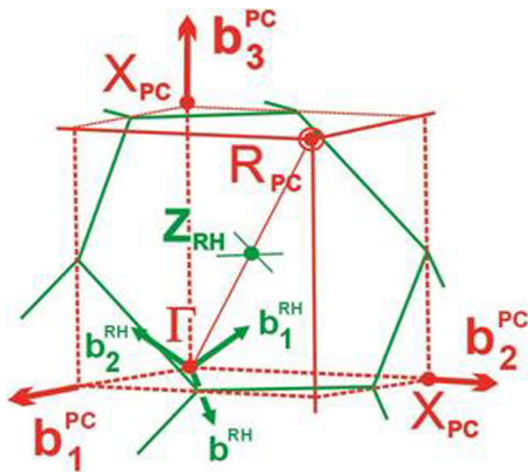


FIG. 6. Embedment of A7 rhombohedral structure (RH) Brillouin zone, in green, to the cubic (pseudo-PC) one, in red; only the irreducible octant is shown. R_{PC} is the PC-RH transformation critical point.

one notes that the intermediate, incommensurate, tetragonal, and monoclinic structures (SbII and SbIV) are very similar in all studied elements [8,36] and structurally interlinked by negligible distortions.

Analyzing the ferrodistorsive ($Z_{PI} = Z_{III}$) transformation between the pseudo-PC and SbIII one finds that a group-subgroup relationship is broken for their space groups $Pm\bar{3}m(Z_{PI} = 1)$ and $Im\bar{3}m(Z_{III} = 1)$. The latter conclusion is also valid for the $R\bar{3}m(2)-Im\bar{3}m(1)$ transformation. A phenomenological scheme for such *reconstructive* phase transitions should contain an intermediate phase whose space group is a common subgroup, normally maximal, for both high-symmetry phases [38]. A relevant geometrically optimal pathway for Sb would occur via antiparallel shifts of alternating $(001)_{BCC}$ atomic planes in the $\pm[110]_{BCC}$ directions. The shifts reduce the crystal symmetry from cubic to orthorhombic $Cmcm(2)$ for general magnitude displacements, then restore it to cubic for special shifts. The corresponding dynamic instability (“soft mode”) develops at the H_{BCC} -point [$\mathbf{k}_{12}^{BCC} = 1/2(\mathbf{b}_1^{BCC} + \mathbf{b}_2^{BCC} - \mathbf{b}_3^{BCC})$], see Fig. 7. However, energy minimization criteria, due to the deviation of the atomic interaction character from the simplest spherical, can produce an alternative pathway and this is the case in Sb. It has been shown in earlier DFT calculations [39] that the complex interplay between electrostatic (Madelung) energy E_{Mad} and the band energy E_{band} stabilizes the pressure controlled structural sequence in Sb: E_{band} governs the structural stability at low

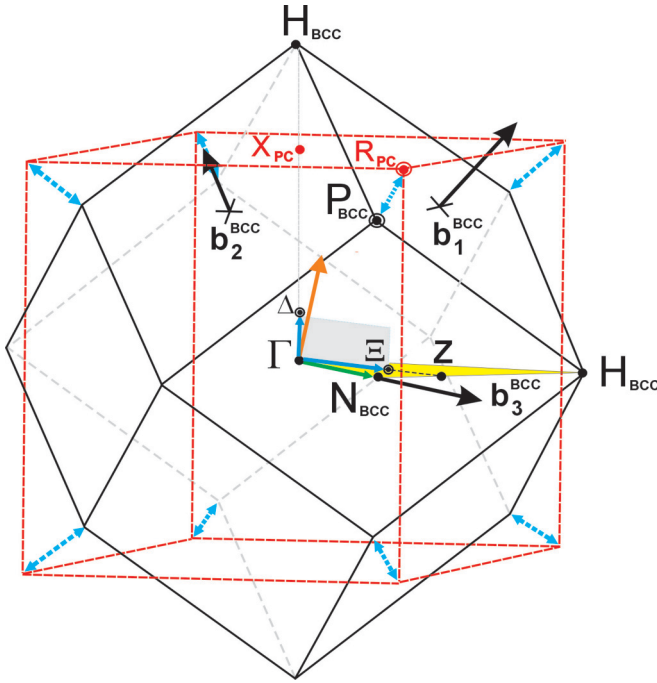


FIG. 7. Brillouin zones of the symmetrized A7 rhombohedral structure, as cubic, and high pressure BCC with the high symmetry points of BCC BZ (continuous black line), and that of the primitive cubic BZ (dashed red line). The $\langle 110 \rangle_R$ phonon dispersion direction coincides with the Γ - N_{BCC} line, the orange arrow shows the $\langle 121 \rangle_R$ phonon dispersion direction. All the critical vectors related to the complex phase transitions lie on the yellow plane Γ - N_{BCC} - H_{BCC} , with the exception of the vector $\mathbf{k}_8$. The two blue arrows correspond to the directions of the critical vectors for the transition SbIII-SbII (SbIV). Dashed double blue arrows indicate a secondary macroscopic BCC-PC lattice distortion (value $\sim 10\%$). The figure in color is available online.

pressure while E_{Mad} dominates at high pressure, stabilizing the densely packed BCC, giving rise to the complex intermediate, incommensurate, structures and directing atoms along a nontrivial pathway from PC to BCC.

Moreover, a universal trend to stabilize intermediate, incommensurate, structures in the vicinity of reconstructive phase transitions was recently predicted in the framework of a more general approach [40]. A model-free, symmetry based theory introduces a universal mechanism for the formation and stabilization of such inhomogeneous states. The mechanism accounts for elastic properties of the crystal lattice via a bilinear coupling between critical displacement gradients

(possible nonuniform order-parameter distribution) and secondary strains. The importance of the latter in reconstructive mechanisms is widely recognized.

Propagation of distortion in the intermediate SbII and SbIV structures is defined by two reciprocal space vectors: (i) the long periodic but commensurate $\mathbf{k}_1 = 2/5(\mathbf{b}_1 - \mathbf{b}_2) - 1/5\mathbf{b}_3$ which lies in the N_{BCC} - Γ - H_{BCC} plane, and (ii) the incommensurate $\mathbf{k}_8 = \mu(\mathbf{b}_1 + \mathbf{b}_2 - \mathbf{b}_3)$, which belongs to the Γ - Δ - H_{BCC} line perpendicular to this plane. Remarkably, the $\mathbf{k}_8^{BCC}$ varies towards the H_{BCC} -point, and BCC (SbIII), therefore, could be considered as a “lock-in” phase or vice versa. Figure 7 and Table I summarize the set of critical points, lines, and planes in the BCC reciprocal space.

d. Discussion

Antimony follows a pathway containing monoclinic and tetragonal incommensurate phases that is favored energetically prior to stabilize the BCC phase. In order to assess the contribution of the lattice dynamics to the pathway towards the BCC phase, we measured the pressure dependence of the phonon dispersions along the $\langle 110 \rangle_R$ direction (green arrow in Fig. 7). This direction is near the BCC-SbII critical vector and in plane with the BCC-pseudo-PC(A7) critical vector. An intense softening is found, with a decrease in energy by the $\sim 15\%$ at 6.5 GPa.

However, the measurements presented in this work, together with previously reported experimental data [9,11], allow us to conclude that antimony shows the presence of another structural mechanism which is a general trend of group Va elements. In fact, the rhombohedral A7 structure seeks to increase its symmetry and stabilize the PC phase. The two main observations supporting this last conclusion are as follows. First, the Raman results on SbI [9,11] revealed simultaneous partial softening of two modes, Γ_1^+ (A_g) and Γ_3^+ (E_g), as one increases pressure. These modes originate from the BZ folding which transfers the critical, triply degenerated, R_5^+ phonon mode from the PC BZ R-node to the A7 BZ center (Γ point, Fig. 6), splitting it into two Raman active modes Γ_1^+ and Γ_3^+ . Second, the pressure dependence of the phonon dispersions along the second direction measured in our study, *i.e.*, in the $\langle 121 \rangle_R$ direction, highlighted as an orange arrow in Fig. 7 in reciprocal space [see also Fig. 5(b) for the phonon dispersion]. This direction lies near the ΓR direction and exhibits a “relative softening” under pressure. In both cases (Raman study and our current results), only partial softening is observed, which is in agreement with the fact that antimony under hydrostatic pressure only initially relaxes

TABLE I. Critical vectors for principal phase transitions.

Phase transition	Critical vector ($\mathbf{k}$)	Point or direction in the BZ ^a
SbIII - SbII (SbIV)	$\mathbf{k}_1^{BCC} = 2/5 (\mathbf{b}_1^{BCC} - \mathbf{b}_2^{BCC}) - 1/5 \mathbf{b}_3^{BCC}$	$\Gamma - \Xi - Z$
	and $\mathbf{k}_8^{BCC} = \mu 2/5 (\mathbf{b}_1^{BCC} - \mathbf{b}_2^{BCC} - \mathbf{b}_3^{BCC})$	and $\Gamma - \Delta - H_{BCC}$
PC - SbI	$\mathbf{k}_{13}^{PC} = 1/2 (\mathbf{b}_1^{PC} + \mathbf{b}_2^{PC} + \mathbf{b}_3^{PC})$	R_{PC}
SbIII - PC	$\mathbf{k}_{12}^{BCC} = 1/2 (\mathbf{b}_1^{BCC} + \mathbf{b}_2^{BCC} - \mathbf{b}_3^{BCC})$	H_{BCC}

^a Direction in the Brillouin zones referred to Fig. 7.

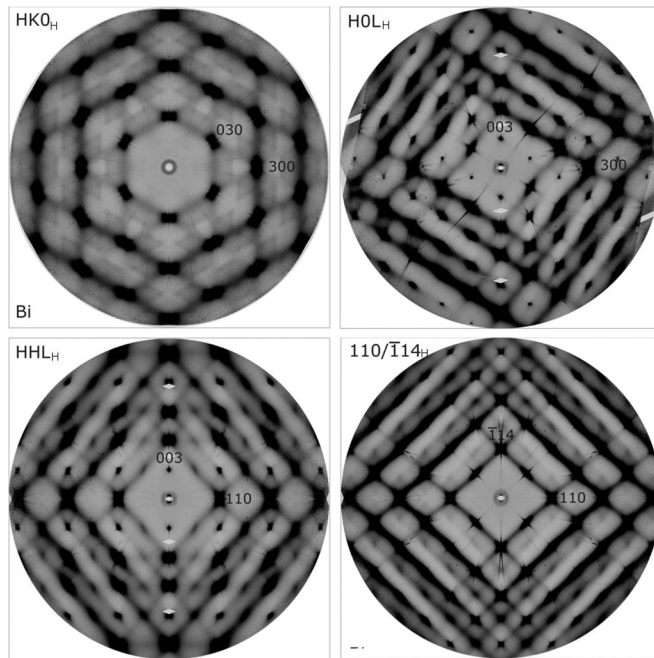


FIG. 8. Diffuse scattering maps of bismuth at ambient conditions. In this case, the entire maps are reconstructed from experimental results. The hexagonal setting is chosen to classify the planes.

the A7 structure and it never reaches the undistorted PC structure.

We conclude our work by reporting the diffuse scattering maps collected on another element of the Va group, bismuth. The measured maps are presented in Fig. 8 and did not reveal any qualitative features different to those of antimony, suggesting a similar pressure trend for this element. Indeed, according to earlier studies, bismuth follows a similar structural path to antimony [1,5,36,41].

V. CONCLUSION

The lattice dynamics of antimony at both ambient and high pressure conditions show anomalies that are understood by means of *ab initio* calculations and phase transition mechanisms analysis. This study focused on the pressure dependence of the phonon dispersions along two critical directions in the reciprocal space corresponding to the canonical Va group phase transitions. The reconstructive phase transition analysis was employed with a focus on those two mechanisms: (I) the A7-PC transition, which describes the relaxation of the rhombohedral distorted structure that is never reached in this particular system and (II) the A7-BCC transition, which passes through two incommensurate structures. In the first case, the relevant phonon dispersion shows a relative softening under pressure, while for the second one, a bigger pressure-induced softening of $\sim 15\%$ is observed along the direction. While both transition mechanisms are mirrored in the phonon dispersions of antimony, the strongest phonon softening is related to A7-BCC transition, which is also the more energetically favorable. The competition of those two mechanisms induces the observed complex behavior of antimony under pressure.

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