Facile synthesis of highly fluorescent free-standing films comprising of graphitic carbon nitride (g-C₃N₄) nanolayers

Ram Manohar Yadav^{a,b} *, Rajesh Kumar^c, Amir Aliyan^d, Pramod S. Dobal ^a, Santoshkumar Biradar^b, Robert Vajtai^{b,e}, Dinesh Pratap Singh^f, Angel A. Martí^{b,d,*} and Pulickel M. Ajayan ^{b,d,*}

^a Department of Physics, VSSD College Kanpur, India-208002

^b Department of Materials Science and NanoEngineering, Rice University, Houston, TX, USA-77005

^c Department of Electrical and Electronic Information Engineering, Toyohashi University of Technology, 1-1 Hibarigaoka, Tempaku-cho, Toyohashi, Aichi, 441-8580, Japan

^d Department of Chemistry, Rice University, Houston, TX, USA-77005

^e Interdisciplinary Excellence Centre, Department of Applied and Environmental Chemistry, University of Szeged, Rerrich Béla tér 1, Szeged, Hungary

^f Department of Physics, University of Santiago, Santiago-9170124, Chile

Abstract

Astounding graphitic carbon nitride (g-C₃N₄) nanostructures have attracted huge attention due to their unique electronic structures, suitable band gap, thermal and chemical stability insinuating as a promising candidate for photocatalytic and energy harvesting applications. Growth of a free standing film is desirable for widespread electronic devices and electrochemical applications. Here we present a facile approach to prepare free-standing films (15mm × 10mm × 0.5mm) comprising of g-C₃N₄ nanolayers by the pyrolysis of dicyandiamide (C₂H₄N₄) utilizing a chemical vapor deposition (CVD) technique. The synthesis is done under low-pressure conditions of Argon (~ 3 Torr) and at a temperature of 600 $^{\circ}$ C. The as-synthesized g-C₃N₄ films are systematically studied for their structural/microstructural characterization by using X-ray diffraction (XRD), scanning & transmission electron microscopy (SEM and TEM), X-ray photoelectron spectroscopy (XPS), Fourier transform infrared spectroscopy (FTIR) and UV-visible spectroscopy techniques. Excitation dependent photoluminescence (PL) spectra of the as-synthesized g-C₃N₄ film exhibit an intense, stable and broad emission peak in the visible region ~ 459 nm. The free standing g-C₃N₄ films emission spectra shows a blue shift and band sharpening as compared to the g-C₃N₄ powder.

Keywords: graphitic carbon nitride film; g-C₃N₄; spray pyrolysis; free-standing; fluorescence; emission

* Corresponding authors: <u>manohar28@gmail.com</u> (Dr. R.M. Yadav) <u>amarti@rice.edu</u> (Prof. A.A. Martí) <u>ajayan@rice.edu</u> (Prof. P.M. Ajayan)

1. Introduction

Ever since the discovery of graphene, two-dimensional (2D) materials have attracted intense scientific attention due to their striking new properties and promising diverse applications in various research areas and in the fabrications of new devices. Among these, graphene-like g-C₃N₄ nanostructures are promising for various applications due to their unique optical and electrical properties¹⁻³. The π -conjugated graphitic planes in g-C₃N₄ structures originate from the carbon and nitrogen atoms linked via sp² hybridization ⁴. The graphene-like layered g-C₃N₄ nanomaterial is composed of two basic structure units: s-triazine and tri-s-triazine (heptazine), and is typically nontoxic and metal-free ⁵⁻⁸. This material exhibits outstanding physical and chemical properties⁵⁻⁸ and is considered one of the best stable allotrope among various carbon nitrides under ambient conditions ⁹. As an organic polymeric semiconductor, g-C₃N₄ has attracted worldwide interest due to a band-gap of ~2.7 eV and its adequate band-positions (delocalized electron states) for various redox reactions ¹⁰. The 2D g-C₃N₄ nanomaterials own good biocompatibility, high fluorescence quantum yield, high chemical stability and good photocatalytic properties ¹¹⁻¹⁷.

The g-C₃N₄ layered materials are usually prepared by a high-temperature pyrolysis process of carbon and nitrogen-rich organic compounds, such as urea (CH₄N₂O) ¹⁸, thiourea (CH₄N₂S)¹⁹, cyanuric chloride (C₃Cl₃N₃)²⁰, dicyandiamide (C₂H₄N₄) ^{21, 22}, ammonium thiocyanate (NH₄SCN) ²³, melamine (C₃H₆N₆) ^{24, 25}, cyanamide (CH₂N₂) ²⁶, and ethylenediamine (C₂H₈N₂) with carbon tetrachloride, etc. Single layered to few layered g-C₃N₄ nanosheets were also synthesized by bulk g-C₃N₄ powder using a top-down approach. Niu et al. ²⁷ reported a controllable thermal oxidation to etch bulk g-C₃N₄ into a few atomic layers that revealed superior photocatalytic activity. For example, Xu et al. ²⁸ developed an H₂SO₄-intercalation method to synthesize single-layered g-C₃N₄ nanosheets and obtained a 3-fold enhancement in photocatalytic H₂ evolution. Also, Zhou et al. ²⁹ reported a low-temperature solid-phase method to synthesize highly fluorescent C₃N₄ dots with a quantum yield of 42% from urea and sodium citrate, and Barman et al. ³⁰ reported a microwave mediated method to prepare highly fluorescent g-C₃N₄ quantum dots with a quantum yield of 29% in formamide.

The g-C₃N₄ can have fascinating applications in many areas of basic science and technology. Particularly, it has significant applications for device fabrication in various fields, such as biosensing, bioimaging, ions detection, photocatalysis, optoelectronic devices, carbon dioxide reduction, water splitting and photodegradation of water pollutants ³¹⁻³⁶. Zhang et al. ³⁷ reported a pathway to prepare ultrathin g-C₃N₄ nanosheets for bioimaging by a liquid exfoliation route from bulk g-C₃N₄. Xu et al. ³⁸ demonstrated the synthesis of ultrathin g-C₃N₄ films supported on attapulgite nanofibers for visible-light photocatalytic activity by degradation of methyl orange. In other applications, Rong et al. ³⁹ developed an effective and facile fluorescence sensing approach for the selective determination of Cr(VI) using g-C₃N₄ nanosheets. Recently, the visible photoluminescence (PL) of carbon nitride bulk and nanopowder have been thoroughly studied. Zhang et al. ¹ reported how the photoluminescence maximum of g-C₃N₄ nanopowder varied with growth parameters and proposed a luminescent mechanism. Zhao et al. ⁴⁰ studied the photoluminescence behavior of functionalized g-C₃N₄ nanosheets grafted with rare earth metal nanoparticles, nonetheless an extensive photoluminescence study of pristine g-C₃N₄ nanolayers and their contrast with the bulk materials is highly desirable to understand the PL phenomenon of these nanomaterials and to make it available for different device applications.

In the present work, we report a simple and one step chemical vapor deposition (CVD) approach to synthesize free-standing $g-C_3N_4$ films. The free-standing $g-C_3N_4$ film was fabricated from thermal condensation of $C_2H_4N_4$ at 600 °C in a low pressure Ar atmosphere. We thoroughly characterized the $g-C_3N_4$ free standing films using spectroscopic techniques (particularly steady-state and time-resolve photoluminescence), XRD, TEM & SEM and further compared this material with the $g-C_3N_4$ powder spectra.

2. Experimental details

2.1. Synthesis of free-standing g-C₃N₄ films and powder

The synthesis of free-standing g-C₃N₄ films was carried out using a pyrolysis-assisted CVD method. In pyrolysis-assisted CVD, the dicyandiamide (C₂H₄N₄) powder was used as C and N source precursor and placed in a quartz tube towards the inlet of Ar gas flow. The outlet of the quartz tube was connected to a vacuum pump to maintain the low pressure (~3 Torr) inside the tube. Due to Ar gas flow with the low vacuum inside the quartz tube, the vapors of C₂H₄N₄ powder enter inside the hot zone of the furnace (600 °C). The condensation of C₂H₄N₄ at high temperature produces the g-C₃N₄ structure deposited inside the quartz tube. After deposition, the furnace was allowed to cool down to room temperature automatically under Ar ambient. A uniform yellow deposition inside the quartz tube at the reaction hot zone was observed (**Fig. 1**). The yellow deposition was collected in the form of a free-standing film which was analyzed further by using

different characterization techniques for structural and optical properties. The $g-C_3N_4$ powder was synthesized using the same precursors of dicyandiamide ($C_2H_4N_4$) at 600 ^{0}C in air atmosphere.

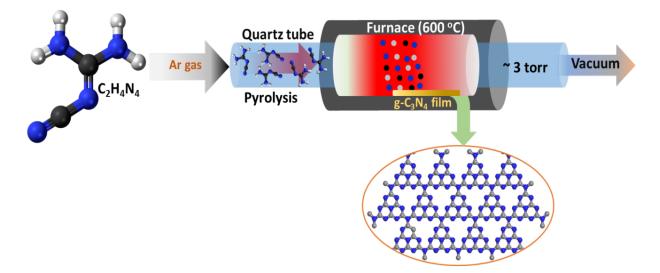


Fig. 1 Schematic representation of pyrolysis-assisted CVD synthesis for free-standing $g-C_3N_4$ films.

2.2. Characterization

The structural/microstructural and optical characterizations were performed via powder XRD, SEM, TEM, XPS, FTIR, UV-visible absorption and emission spectroscopy. The morphological and structural analyses of the synthesized samples were determined by SEM (FEI Quanta 400 FEG ESEM), TEM (JEOL 2100 FEG TEM) and XRD (Rigaku Smart Lab XRD). FTIR spectra were recorded by using a Nicolet FTIR Infrared Microscope. XPS system from PHI Quantera XPS (on a PHI-5000C ESCA system with Al K α) was used for stoichiometric analysis. UV-visible absorption spectra were obtained using Shimadzu UV-2450 spectrophotometer. Steady-state fluorescence experiments were performed in a HORIBA Jovin Yvon Fluorolog 3 fluorometer and time-resolved experiments in an Edinburgh Instruments OD470 single-photoncounting spectrometer with a 372 nm picosecond pulse diode laser and a high-speed red detector.

3. Results and discussion

Fig. 2a shows the optical image of free-standing $g-C_3N_4$ film of approximate area ~150 mm² with the thickness about 0.5 mm (*l=15mm; w=10mm; t=0.5mm*). Figs. 2b-e show the SEM images of $g-C_3N_4$ film at different magnifications. As shown in Fig. 2b and Fig. 2c, the synthesized $g-C_3N_4$ film exhibited the nanosheet structures with top surfaces containing porous morphology with

wrinkled flakes. The high magnification images (Fig. 2d and Fig. 2e) show the nanoporous structures of $g-C_3N_4$ film, as marked by arrows.

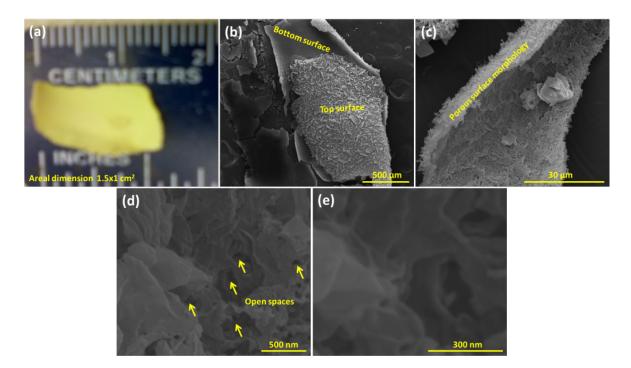


Fig. 2 Optical image (a) and SEM images (b-e) of as grown free-standingg-C₃N₄ films. SEM images are taken at different magnifications.

The detailed morphology of the free-standing g-C₃N₄ films was also investigated by TEM analysis (**Fig. 3a-c**), which confirmed their 2D structure and low thicknesses up to ~3nm. The g-C₃N₄ nanolayers exhibited sheets-like morphology with lengths up to several nanometers and of variable thicknesses as shown by arrows (**Fig. 3a**). In order to further examine the structure, high-resolution TEM (HRTEM) analysis was employed. Fig. 3b clearly revealed that the thickness of the g-C₃N₄ nanolayers is non-uniform. The different number of layers can be predicted due to the contrast variation (separated by dotted regions) and indicated by thin and thick nanolayers in **Fig. 3b**. It can be expected that thick nanolayers structure are formed due to the folding of g-C₃N₄ nanolayers.

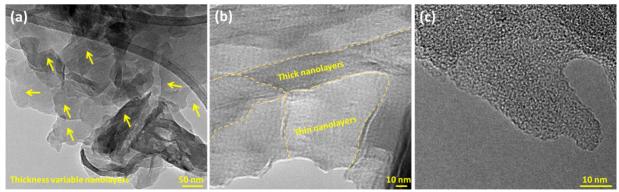


Fig. 3 TEM (a) and HRTEM (b,c) images of g-C₃N₄ film.

In order to investigate the crystalline structure of the free-standing g-C₃N₄ films, the XRD pattern of g-C₃N₄ film was analyzed. The observed XRD pattern of the g-C₃N₄ film, as shown in **Fig. 4a**, displays two diffraction peaks at 12.24° and 27.24° that correspond to the (100) and (002) planes of g-C₃N₄ respectively. This indicate the graphite-like inter-planar stacking of the conjugated aromatic units of CN that are the characteristic stacking structures of graphitic-like materials (hexagonal g-carbon nitride) (JCPDS card no. 87-1526) ^{41, 42}. We observed the same strong diffraction peaks from all the samples prepared, which indicates that free-standing g-C₃N₄ films with stable crystalline structure were successfully prepared from the pyrolysis of C₂H₄N₄.

In addition to X-ray diffraction analysis, FT-IR spectroscopy was used to characterize the chemical structure. A typical FT-IR spectrum of the free-standing g-C₃N₄ film is shown in **Fig. 4b**. Several bands are displayed in the region of 4000-700 cm⁻¹, which are characteristic of the C-N stretching. The sharp and intense transmittance band at 817 cm⁻¹ can be attributed to the vibration of triazine rings, indicating the existence of the units with -NH and -NH₂ groups ⁴³. The several peaks in the region from 1200 to 1600 cm⁻¹ can be attributed to typical stretching modes of either trigonal C-N (-C)-C or bridging C-NH-C units in g-C₃N₄ film ^{44, 45}. The broad peaks between 3070 and 3285 cm⁻¹ correspond to N-H and O-H stretching and hydrogen-bonding interactions ^{32, 44}.

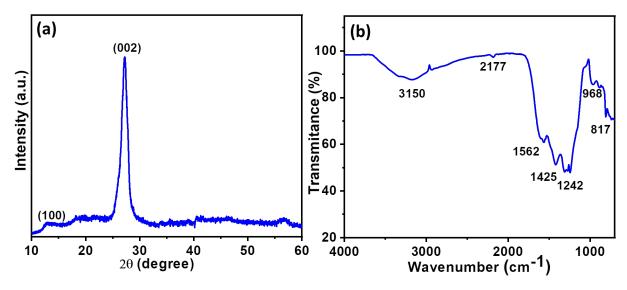


Fig. 4 (a) XRD pattern and (b) FT-IR spectra g-C₃N₄ film.

The XPS technique is well recognized and has been extensively utilized to provide binding energy of core level electrons of atoms in the solid. **Fig. 5a**, shows the C1s spectrum of a free-standing g-C₃N₄ film, which shows two broad peaks. These two deconvoluted peaks at 284.9 and 288.2 eV correspond to sp² C-C and sp² N-C=N bonds, respectively. The high-resolution N1s spectrum as shown in **Fig. 5b** has been deconvoluted into three peaks at 398.6, 399.9 and 401.0 eV, which were assigned to the sp²-hybridized N (C-N-C), sp³-hybridized N (tertiary N, (N-[C]₃) and amino functional groups with H atom (C-N-H), in that order ^{5, 43, 46-48}. These XPS results confirm that the basic substructure of the synthesized material is the heptazine heterocyclic ring unit.

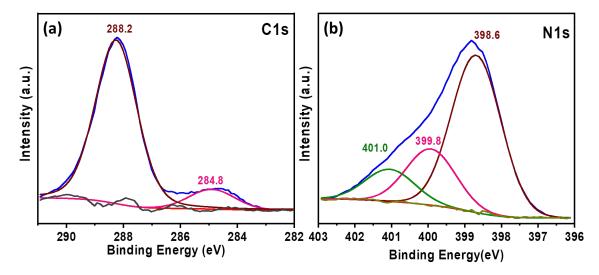


Fig. 5 High-resolution (a) C1s and (b) N1s XPS spectra of g-C₃N₄ film.

Fig. 6 displays the diffuse reflectance and fluorescence spectra of as-synthesized freestanding $g-C_3N_4$ film and $g-C_3N_4$ powder. Fig. 6a shows the diffuse reflectance spectrum of $g-C_3N_4$ film and $g-C_3N_4$ powder. The $g-C_3N_4$ film absorbed visible light up to ~470 nm, suggesting that it could be activated by visible light which is favorable for bio-imaging. ^{49, 50} Diffuse reflectance spectra has been commonly used as standard technique to estimate the band gap of semiconducting materials ^{51,52, 53}. The band gap for the samples can be calculated according to the following relation

$$F(R)hv = A(hv - E_g)^{\frac{n}{2}} \dots 1$$

Where $F(R) = (1-R)^{\frac{2}{2}R} \dots 2$
• $F(R)$ is the Kubelka–Munk (K–M) function.

In the equation, *n* equals to 1 or 4, depending on whether the transition is direct or indirect, respectively. X. Wang et al⁵⁴ showed that g-C₃N₄ has a non-isotropic band structure with a direct bandgap at the Γ point and only dispersion along the Γ -X direction parallel to the chain. There are reports suggesting direct⁵ as well as indirect¹⁶ band gap for g-C₃N₄. Here we are discussing the direct band gap although we have calculated the direct as well as indirect band gap for g-C₃N₄ films and powder (Fig. S1 a, b). The values of E_g for free-standing g-C₃N₄ film and g-C₃N₄ powder are calculated to be 2.86 and 2.70 eV, respectively (Fig. S1 a, b). The blue shift of the absorption edge and increasing of E_g value for g-C₃N₄ film as compared to that of g-C₃N₄ powder could be ascribed to the quantum confinement effects ^{21, 27, 55}. Fig. 6b shows the fluorescence spectra of the g-C₃N₄ film and g-C₃N₄ films contain relatively uniform particle size distribution as compared to that in g-C₃N₄ powder. This clearly suggests that g-C₃N₄ films contain relatively uniform particle size distribution as compared to that in g-C₃N₄ powder ²⁹.

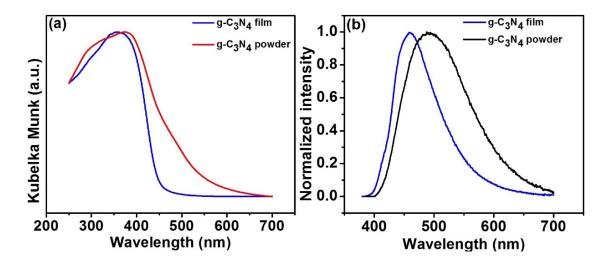


Fig. 6 (a) Diffuse reflectance spectra, and (b) PL spectra of g-C₃N₄ film and g-C₃N₄ powder.

Fig. 7a presents the excitation-dependent PL spectrum of free-standing g-C₃N₄ films. Various excitation wavelengths in the range of 300-424 nm were used to study the fluorescence emission spectrum. As expected, the fluorescent spectrum shows that the emission band (~ 450-460 nm) remains unchanged for all excitation wavelengths. Nonetheless, the PL emission peak intensity varies with excitation wavelength as shown in **Fig. 7b**. It was found to increase with increasing the excitation wavelength with maximum intensity at 372 nm. The constant emission wavelength band (450-460 nm) from all the synthesized free-standing g-C₃N₄ films indicate that the g-C₃N₄ films are optically similar. Also, the films show high photostability, with the fluorescence intensity remaining constant after continuous irradiation.

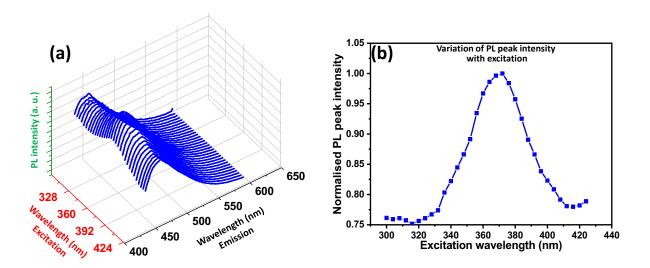


Fig. 7 (a) Excitation-dependent fluorescence emission spectrum (excitation wavelengths changes from 300 nm to 424 nm) and (b) variation of emission intensity of $g-C_3N_4$ film with excitation wavelength at emission 459 nm.

To compare the results with g-C₃N₄ film, we have also studied the PL emission spectra of g-C₃N₄ powder. **Fig. S2** shows the excitation dependent PL emission spectra of g-C₃N₄ powder. The maximum PL intensity was found around 488 nm with excitation wavelength 330 nm. The PL emission profiles of g-C₃N₄ powder and g-C₃N₄ film are shown in **Fig. S3**. The emission profiles clearly illustrate the difference in emission from g-C₃N₄ powder and g-C₃N₄ film. This plot reveals that the emission profile of g-C₃N₄ film is relatively narrow (**Fig. S3(a)**), while in the case of g-C₃N₄ powder the band is broader (**Fig. S3(b)**). Overall analysis obtained from the absorption and emission in the visible region, suggests that free-standing g-C₃N₄ film can be used for a variety of applications from solar energy conversion to bio-imaging applications.

Fig. 8 shows the fluorescence decay curve of the g-C₃N₄ film using 372 nm excitation wavelength with instrument response function (IRF) and multi exponential fit curve. The decay curve clearly shows the deviation from the single-exponential decay, which indicates that multiple environments are represented in the emission profile ⁵⁶. By the multi-exponential fitting, the fluorescence lifetimes of the free-standing g-C₃N₄ film were found to be about 27 ns, 120 ns and 90 ns. The nanosecond lifetime of g-C₃N₄ films endows their potential applications in bio-imaging^{37, 57} displays, lighting⁵⁸, etc. Bayan et al.⁵⁸ reported the white luminescence from heterojunctions based on two dimensional (2D) graphitic carbon nitride (g-C₃N₄) nanosheets and ZnO nanorods and shows the applicability of g-C₃N₄ film and reported their excellent performance, stability and recyclability for the catalytic dye degradation. Xie et al.⁶⁰ reported the in- situ growth of g-C₃N₄ films on transparent substrate by solvothermal method. Their results revealed that g-C₃N₄ films are convenient for photo electrochemical energy conversion

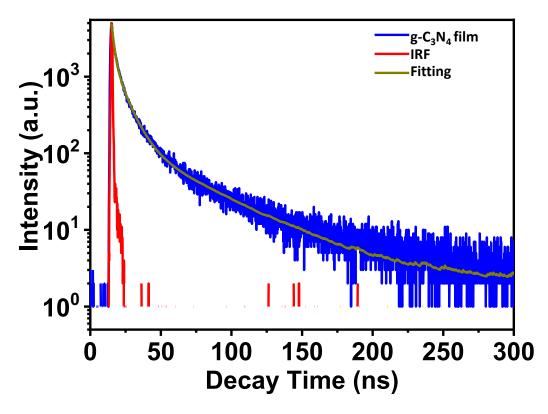


Fig. 8 Decay curve of g-C₃N₄ free standing film with multi exponential fitting and IRF curve.

Fig. 9 shows the transient PL spectra of $g-C_3N_4$ powder and $g-C_3N_4$ films. The emission intensities of these peaks exponentially decrease with increasing the decay time. The transient PL emission spectra of $g-C_3N_4$ film (Fig. 9a) exactly correlates with the finding of excitation-dependent fluorescence emission spectrum (Fig. 7) however transient PL emission spectra of $g-C_3N_4$ powder (Fig. 9b) reveal two peaks at 480 nm and 525 nm as compared to one broad peak of fluorescence emission spectrum at 489 nm (Fig. S2 & S3b). The reason of two peaks in transient spectra is not clear but Zhang et al¹ suggested that the obvious difference between the steady-state and transient PL emission spectra implies that there is more than one PL emission center during the emission process of carbon nitride powder and it seems that these two peaks merged in one peak in PL spectra of $g-C_3N_4$ powder. The PL emission spectra of the $g-C_3N_4$ film show the single sharper peak as compared to $g-C_3N_4$ powder. It is clearly evident that during emission process more PL emission centers are present in $g-C_3N_4$ powder as compared to $g-C_3N_4$ films.

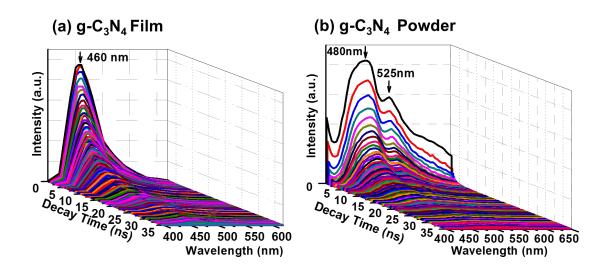


Fig. 9 Time resolved PL emission spectra of (a) g-C₃N₄ film and (b) g-C₃N₄ powder.

The blue shift for PL emission in g-C₃N₄ films could be attributed to quantum confinement effects arising due to smaller particle size in g-C₃N₄ films as compared to powder initiating the band gap to widen up ²⁸. It is evident from the XRD and XPS spectra, the g-C₃N₄ free standing films consisted of heptazine structure⁶¹ which has higher disorder and delocalization of electrons. Zhang et al. ¹ revealed that the delocalization of electrons and thus overlap of orbitals leads to vanishing or weaken the transition probability between Π and Π^* states ^{62, 63} and thus the peaks at 525 nm disappears in g-C₃N₄ films.

4. Conclusions

In summary, we have developed a facile one-step CVD assisted approach for the synthesis of high-quality and large area (15mm×10mm) free-standing g-C₃N₄ films by direct pyrolysis of C₂H₄N₄ in a quartz tube at 600 ⁰C. The film was made up of g-C₃N₄ nanolayers with variable thickness due to folding and overlapping of various nanolayers. The detailed PL spectra and decay analysis obtained with different excitation energies revealed no significant change in the position of the emission band and hence confirms the existence of a particular energy band. The assynthesized g-C₃N₄ film exhibited stable and strong photoluminescence emission centered around 455-460 nm. The photoluminescence lifetime analysis as detected from the time decay spectra suggest this free-standing g-C₃N₄ film as a prospective candidate for bioimaging and display applications. This simple approach can be expanded for the synthesis and creation of stable polymeric g-C₃N₄ film with various microstructures.

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