

# Bottom-up construction of electrochemically active living filters: from graphene oxide mediated formation of bacterial cables to 3D assembly of hierarchical architectures

Pu Deng<sup>1,2,†</sup>, Weiqin Sheng<sup>1,3,†</sup>, Andrew Xu<sup>1,†</sup>, Yixin Zhang<sup>1</sup>, Xiaochuan Dai<sup>1</sup>, Richard Vo<sup>1</sup>, Huan-Hsuan Hsu<sup>1\*</sup>, Xiaocheng Jiang<sup>1\*</sup>

<sup>1</sup>Department of Biomedical Engineering, Tufts University, Medford, Massachusetts 02155, USA

<sup>2</sup>Thomas Lord Department of Mechanical Engineering & Materials Science, Duke University, Durham, North Carolina 27705, USA

<sup>3</sup>School of Electronic Information, Hangzhou Dianzi University, Hangzhou 310018, People's Republic of China

**ABSTRACT:** Living composites comprising of both biotic and abiotic modules are shifting the paradigm of materials science, yet challenges remain in effectively converging their distinctive structural and functional attributes. Here we present a bottom-up hybridization strategy to construct functionally coherent, electrochemically active biohybrids with optimal mass/charge transport, mechanical integrity, and biocatalytic performance. This biohybrid can overcome several key limitations of traditional biocarrier designs and demonstrate superior efficiency in metabolizing low-concentration toxic ions with minimal environmental impact. Overall, this work exemplifies a new bio-integration strategy that complements existing synthetic biology toolsets to further expand the range of material attributes and functionalities.

**KEYWORDS:** Exoelectrogen, living hybrid materials, 3D assembly, hierarchical structures, bioelectrochemical systems.

1 Exoelectrogens, microorganisms that are capable of cross- 39  
2 membrane transport of metabolically generated electrons,<sup>1</sup> rep- 40  
3 resent unique living building blocks for the construction of elec- 41  
4 troactive biosystems. Encoded with a range of extracellular 42  
5 electron transfer (EET) pathways,<sup>2</sup> they have been expanding 43  
6 the spectrum of functional living material designs and empow- 44  
7 ering applications in bioelectronics,<sup>3</sup> energy,<sup>4</sup> and environmen- 45  
8 tal science.<sup>5</sup> The performance of these living devices to date, 46  
9 however, remains limited by the compromised charge and/or 47  
10 mass transport that is inherent to the native structural assemblies 48  
11 (*i.e.* electroactive biofilms). For example, modeling of *G. sul-* 49  
12 *furreducens* biofilm grown on planar electrodes indicates that 50  
13 the cells furthest from the electrode barely respire/divide due to 51  
14 their low efficiency to remotely exchange electrons with elec- 52  
15 trode,<sup>6</sup> while the high diffusion resistance across the native bio- 53  
16 film further limits the access to nutrient/metabolite exchange 54  
17 for inner layers of cells and their electrochemical performance.<sup>7</sup> 55

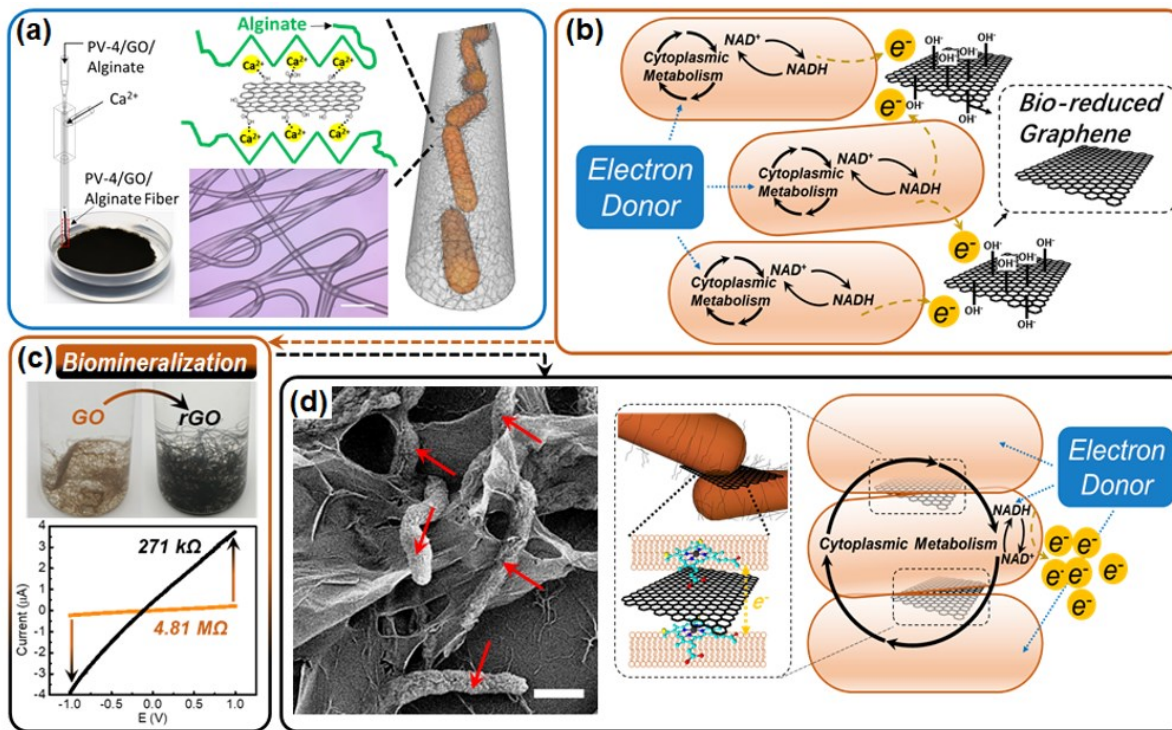
18 Bio-hybridization with complementary material components 56  
19 could potentially overcome these intrinsic limits imposed by bi- 57  
20 omolecular/cellular building blocks. Through mixing with con- 58  
21 ductive dopants, such as metal nanoparticles,<sup>8</sup> carbon nano- 59  
22 tubes,<sup>9</sup> graphene,<sup>10</sup> or conducting polymers,<sup>11</sup> composite bio- 60  
23 films have been fabricated with increased electrical conductiv- 61  
24 ity and mechanical strength, enabling enhanced power genera- 62  
25 tion,<sup>12</sup> biosynthesis,<sup>13</sup> or wastewater treatment.<sup>14</sup> These physi- 63  
26 cal- or chemical-based hybridization approaches, however, are 64  
27 typically limited by the inefficient structural (poor interfacial 65  
28 contact) and functional (large interfacial charge transfer barrier) 66  
29 integration at the biotic-abiotic junctions. Additionally, the lack 67  
30 of microenvironmental control leads to substantial fluctuation 68  
31 in cell functioning (*e.g.* metabolic rate) over longer length and 69  
32 time scales that further impedes system-level performance.<sup>15</sup> 70

33 In comparison, biologically driven hybridization processes, 71  
34 such as bio-mineralization, present unique opportunities for im- 72  
35 proved cross-system integration by initiating and directing the 73  
36 interplay with foreign components from the bottom up.<sup>16</sup> Bone 74  
37 mineralization, for example, enables highly specific and or- 75  
38 dered incorporation of nanocrystalline apatite in the gap regions

of self-assembled type I collagen fibrils, leading to hierarchi-  
cally structured calcified tissues with superior mechanical prop-  
erties that cannot be matched by synthetic materials. Similarly,  
exoelectrogens are known for their unique capability to electri-  
cally interact with external electron acceptors and in-situ de-  
posit inorganic materials (such as Au, Ag, Pd, and FeS<sub>x</sub> nano-  
particles) on outer-membranes.<sup>17</sup> These dissimilatory reduction  
processes actively engage direct electron exchanges with bio-  
synthesis to bridge biological and solid-state charge transport  
systems at these hybrid interfaces.

Here we report a multiscale hybridization strategy that con-  
verges both assembly modalities to generate electrochemically  
active living materials with enhanced performance. In particu-  
lar, the living catalysts, *Shewanella loihica* PV-4, are converted  
into conductive bacterial “cables” through spatially-confined  
biomineralization, where the metabolic pathways of individual  
bacterium are electrically conjugated to form expanded EET  
networks with significantly increased active surface area. These  
1D composite building blocks are further engineered into 3D  
hierarchical frameworks through bio-printing with rationally  
designed physiochemical characteristics (Figure 1a). As-pre-  
pared bio-hybrids have the potential to overcome the inherent  
limits in mass/charge transport of native bioelectrochemical  
systems, reinforce a range of applications in energy and envi-  
ronmental science, and broadly impact how functional living  
materials can be designed and constructed.<sup>18</sup>

At the cellular level, structural and functional integration is  
initiated with the bio-reduction of graphene oxide (GO), the  
critical “linker” that is converted into conductive rGO through  
the metabolism of *S. loihica*. Based on our previous research,  
GO holding intimate contacts with bacteria can be fully con-  
verted to reduced GO as confirmed by Raman spectroscopy.<sup>19</sup>  
These bio-reduced GO can intimately couple with its outer  
membrane cytochromes and electrically connecting adjacent  
cells (Figure 1b). Specifically, bioinks formulated with *S.*  
*loihica*, GO and alginate are processed and extruded through a  
co-axial microfluidic device,<sup>20</sup> where the ionically crosslinked

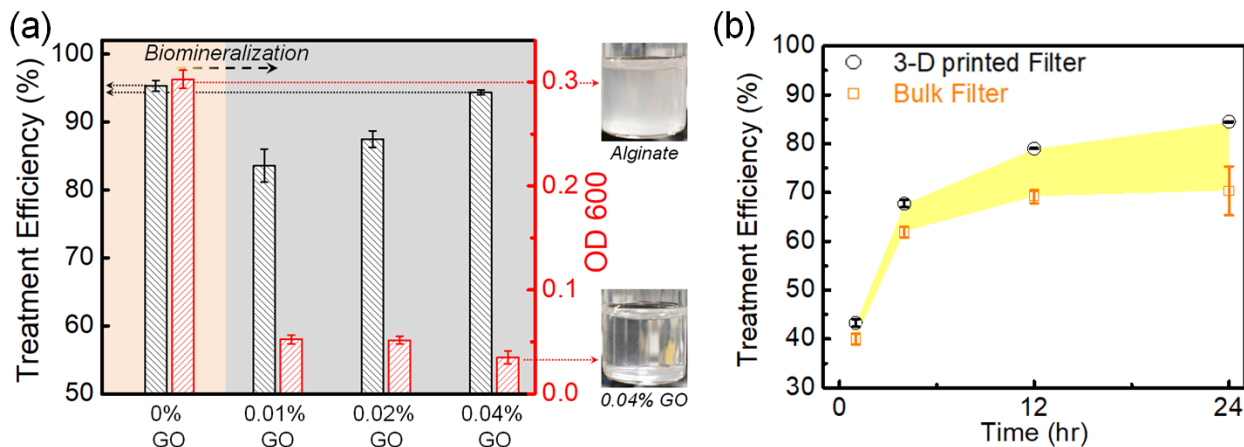


**Figure 1.** Design, assembly and characterization of the living filter: (a) schematics showing the living filter design at ensemble (3-D assembly of the microfibers) and microscopic (graphene oxide (GO)-cation-alginate interactions) levels; (b) Schematic showing the dissimilatory reduction by alginate encapsulated *S. Ioihica* before biomineralization; (c) Biomineralization/bioreduction of GO, which is evidenced by both the brown-to-black color change (top) and the decrease of electrical resistance of bacterial cables (bottom); (d) Schematic and SEM characterization of assembled/biomineralized cell network (scale bar: 300 nm), of which the individual *S. Ioihica* are structurally- and functionally- integrated through the reduced GO enabled cellular assembling.

alginate shell provides porous scaffolding with structural similarity to polysaccharides for 1D cell confinement, while allowing efficient nutrient/metabolite exchange. The bio-transformation of GO into rGO is evidenced by the progressive color change (brown to black) of the hydrogel fibers during a 24-hour culture (Figure 1a) as well as 95% reduction of two-probe resistance (Figure 1c). As-reduced GO represents bio-synthetic electron conduits that actively “wire” the redox center of cytochromes in electroactive conformations and significantly reduces the barrier for interfacial charge transport.<sup>19</sup> This GO-hybridized fiber allows significantly expanding the effective EET active surface area that is critical to many downstream applications (e.g. microbial fuel cells and bio-remediation). At the molecular level, the structural integrity of the composite microfibers is further strengthened by the interaction between GO and alginate (Figure 1a), where the carbonyl and hydroxyl groups on GO help confine the crosslinking cations within the alginate matrices.<sup>21</sup> The chronic stability of the biohybrid is significantly improved, which remains intact and robust after > 48 hours, while alginate/*S. Ioihica* fibers without GO becomes fragile due to the degradation/dissociation of the cross-linked alginate and weaker cell-cell interconnections (Figure S1 & S2). As a result, mechanically reinforced and electrochemically coherent bacterial networks can be formed (Figure 1d and SEM insert).

The hierarchical structural/functional integration at larger scales is achieved through spatially-organized assembly of these 1D hybrid building blocks via bioprinting. A customized, extrusion-based bioprinter (Figure S3) with a multi-inlet microfluidic nozzle is developed, where factors defining individual

fibers (diameter and composition) and their 3D placement (pitch/porosity) can be programmed (Figure S4&S5). This approach allows the bottom-up engineering of key mass- and charge-transport properties that are critical to many downstream applications. For example, biological water treatment provides an attractive opportunity in effective toxic ion removals, which can eliminate the disadvantages of the physical and chemical based treatment techniques.<sup>12</sup> Recently, many biological species such as algae, bacteria, fungi are studied toward this prospective. Among all these species, the unique EET process allows EABs to directly reduce the highly toxic soluble inorganic ions (i.e. Chromium,<sup>22</sup> Arsenic,<sup>23</sup> Cadmium,<sup>24</sup> etc.) toward solid-state low-toxicity nanoparticles. Additionally, since wastewater is abundant in organic matters as the electron donors, EABs are self-sustainable in this environment and have demonstrated extraordinary possibilities in electricity recovery during water treatments.<sup>25</sup> Hence, EABs offer exclusive possibilities in the development of next generation low cost, low energy consumption and self-sustainable wastewater treatment strategies.<sup>26</sup> As a proof of concept, we demonstrate the design and application of this 3-D biocatalyst network as a “living filter” for bio-remediation, which converge the advantages of existing treatment options, i.e., (a) biocarriers (low biomass contamination but compromised mass/charge transport); and (b) planktonic bacteria (high treatment efficiency but requiring onerous post-treatment). Their performance is systematically evaluated in simulated wastewater treatments along with their bulk- or planktonic- counterparts for the reduction of hexavalent chromium (Cr(VI)), one of the most common sources of water



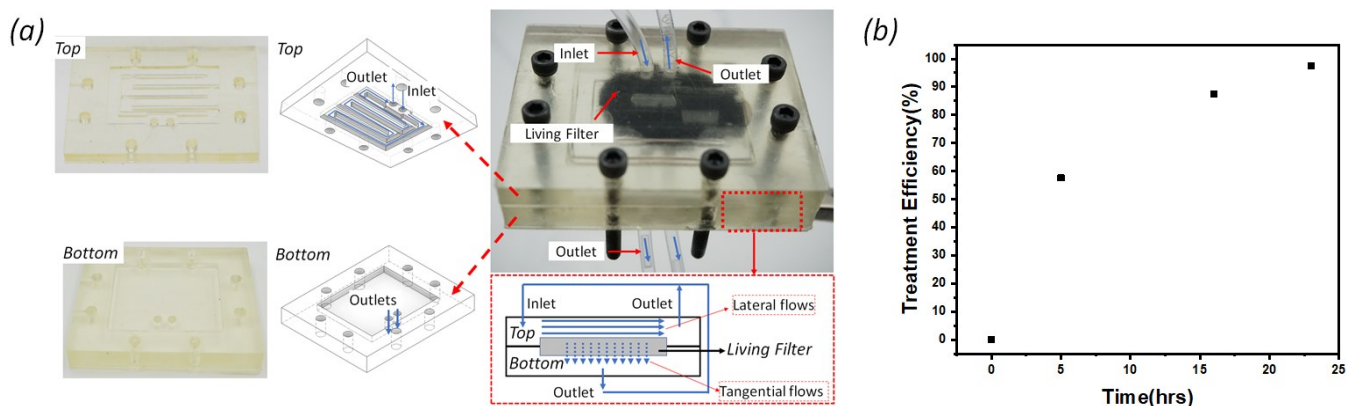
**Figure 2.** Chronic stability and performance of living filters with different compositions and macro-structures: (a) Cr(VI) treatment efficiency and planktonic/leaked *S. loihica* level (OD600, red bars) of filters made of 5 mL bioink with different GO contents. The high efficiency of filter made of pure alginate (0% GO) is mainly attributed to the planktonic *S. loihica*; (an insert) Images of treated water with high (0.3) and low (0.03) OD600; (b) Cr(VI) treatment efficiencies of mesh (60% macro-porosity) versus solid (0% macro-porosity) filters made of 2 mL bioink. Yellow area indicates the difference in treatment efficiency between two types of filters (n = 3).

1 contaminants in U.S.<sup>22</sup>. The procedures of simulated 40  
 2 wastewater treatments are detailed in support information. The 41  
 3 initial concentration of Cr<sup>6+</sup> (190 μM) were selected based on 42  
 4 previous research [doi:10.1128/AEM.02463-10] 43  
 5 The incorporation and biomineralization with GO are found 44  
 6 to play important roles in improving the structural integrity and 45  
 7 treatment efficiency of the living filters. As shown in Figure 2a, 46  
 8 the OD-600 (red bars, correspondent to the planktonic *S.* 47  
 9 *loihica*) of the post-treatment solution by filters made of pure 48  
 10 alginate can reach 0.3 after 24 hours, almost 20% of the fed- 49  
 11 batch (planktonic) culture in the stationary phase (1.5-2.2). This 50  
 12 result indicates that the native bacteria/alginate composites are 51  
 13 prone to dissociate and leak, particularly with the mesh design, 52  
 14 as a result of its significantly increased surface area/decreased 53  
 15 alginate thickness when compared with conventional biocarriers. 54  
 16 However, biomineralization can reinforce the intercellular 55  
 17 connections by “gluing” adjacent *S.loihica* with rGO to im- 56  
 18 prove chronic stability. Together with the interactions between 57  
 19 alginate and GO that stabilize the crosslinks of alginate mole- 58  
 20 cules, the OD-600 can be maintained at below 0.07 during the 59  
 21 treatment with GO-doped living filters (Figure 2a). Overall, 60  
 22 these results demonstrate significantly improved mechanical in- 61  
 23 tegritiy and chronic stability of living filters with the incorpor- 62  
 24 ation of GO linkers, hence minimal biomass production/post- 63  
 25 processing, which remains one of the major challenges in bio- 64  
 26 logical water treatment. 65  
 27 The hybridization with GO also enhances the overall treat- 66  
 28 ment efficiency by improving the system-level EET (Figure 2). 67  
 29 The treatment efficiency of Cr(VI) is determined by a EPA ap- 68  
 30 proved standard method detailed in support information. As 69  
 31 shown, the reduction of Cr(VI) is positively correlated with the 70  
 32 increase of GO concentration, and reaches 92% at GO 0.02%. 71  
 33 This increased biocatalytic kinetics can be attributed to the rGO 72  
 34 mediated electrical interconnection between encapsulated *S.* 73  
 35 *loihica*, which substantially expands the active EET area be- 74  
 36 yond the outer membranes of individual bacterium (Figure 1d). 75  
 37 Although filters made of pure alginate show a higher removal 76  
 38 rates at 96%, this is mainly contributed by the leaked planktonic  
 39 *S. loihica* (OD 600 = 0.3 as compared with 0.03 for GO doped

filters, Figure 2a inset), which are immediately accessible to nu-  
 trients/Cr(VI) and metabolically more active. In GO-doped liv-  
 ing filters, despite the mass transport barrier imposed by the al-  
 ginate/GO matrix (to maintain minimal bacteria leakage), the  
 Cr(VI) removal remains competitively efficient as a result of  
 improved charge and mass transport compared with traditional  
 biocarriers. Overall, the experiments demonstrated that bio-re-  
 duced GO is critical in the biohybrid network, which not only  
 maintain the biocatalytic/electron transport efficiency toward  
 the comparable level as planktonic bacteria (which are much  
 more mass transport favorable), but also maintain the long-term  
 structural integrity to avoid the bacteria leakage. Besides, the  
 bio-reduced GO enhanced structural integrity provides addi-  
 tional possibility for the recycle/reusage of living filter toward  
 the life span of *S. loihica*.<sup>27</sup> While further increasing GO may  
 provide additional benefit for treatment efficiency enhance-  
 ment, the higher GO content can also challenge the printability  
 (forming aggregates that could block the printing nozzle) and  
 long-term stability (incompatibility of hydrophobic rGO with  
 alginate hydrogel that leads to brittle fibers) of living filter.<sup>28</sup>

To further evaluate the effect of filter structure on mass  
 transport and Cr(VI) reduction, we compare the performance of  
 assembled mesh filter (60% macro-porosity) with its bulk coun-  
 terpart (0% macro-porosity) made of identical components  
 (Figure 2b) (SI Methods). The difference in treatment effi-  
 ciency between the two filter designs is highlighted in yellow.  
 Rapid Cr(VI) conversion can be achieved by both mesh- (68 %) and  
 bulk- (62 %) filters within the first 4-hour, when the initial  
 high concentration gradient of Cr(VI) drives efficient inward  
 diffusion. Cr(VI) treatment by bulk filter plateaus after 12 hours  
 when diffusion becomes a limiting factor (Cr(VI) at ~ 67 μM,  
 ~65% treated). In contrast, the Cr(VI) treatment steadily in-  
 creases with mesh filter design until the experiments are termi-  
 nated at 24 hours. The 24-hour duration is selected while the  
 amount/effect of planktonic *S. loihica* remain negligible (OD600  
 = 0.03). The enhanced mass transport can be attributed to the  
 thin fiber diameter (~23 μm) and hierarchically assembled 3D





**Figure 3.** Living filter performance under continuous flow setting: (a) images and schemes of the bioreactor. In the reactor, the top layer was the flow channel that introduced both lateral and tangential flows to the living filter, the middle layer was the living filter and the bottom layer was the reservoir for collecting the tangential flow to be delivered back to main reservoir; and (b) the 24-hour Cr(VI) treatment efficiency of the living filter inside the bioreactor.

1 porous structure, which becomes critically important in over-  
 2 coming the diffusion limit in bulk alginate (pore size  $\sim 5$  nm,  
 3 diffusion coefficient  $4.10 \times 10^{-6}$  cm<sup>2</sup>/min), especially when the  
 4 concentration gradient is low.<sup>29,30</sup>

5 The structural integrity and mechanical stability of free-  
 6 standing living filters could significantly reduce the cost and  
 7 energy consumption associated with the integration, mainte-  
 8 nance, and post-treatment of the biocatalysts. The design is also  
 9 fully compatible with existing wastewater treatment infrastruc-  
 10 ture for sustainable operation. In particular, we have designed  
 11 a semi-continuous flow bioreactor in which both lateral and tan-  
 12 gential flows can be programmed to maximize the flow-filter  
 13 interaction while reducing the direct hydrodynamic impacts  
 14 (**Method & Figure 3**). In the test, Cr(VI) conversion rate was  
 15 significantly increased (60% at 4 hours, and 100% at 24 hrs) as  
 16 compared with the results from batch processing (Figure 2a &  
 17 3), demonstrating the potential of dynamic flow programming  
 18 to further circumvent native mass transport limits at low con-  
 19 centration regimes.

20 In summary, we demonstrate a multi-modality, bottom-up as-  
 21 sembling strategy to generate electrochemically active living  
 22 filters with seamless structural and functional integration, of  
 23 which, assembled *S. loihica* networks have demonstrated their  
 24 unique biocatalytic ability for Cr(VI) removal while their effi-  
 25 ciency and chronic stability are enhanced significantly through  
 26 the GO-assisted assembling processes. At the cellular level, the  
 27 biomineralization/hybridization with GO inside hydrogel fibers  
 28 leads to improved EET and strengthened inter-cellular/molecu-  
 29 lar interactions that eventually benefit the mechanical stability  
 30 and performance enhancement. At the macroscopic level, these  
 31 biohybrid cables are hierarchically assembled into a 3-D biocat-  
 32 alytic network that largely overcomes the mass transport limit  
 33 in traditional biocarrier design. Moving forward, the design and  
 34 implementation of electrochemically active biohybrids will add  
 35 another important dimension to the existing library of func-  
 36 tional living materials, and open up extensive opportunities in  
 37 accomplishing widely-defined tasks (e.g. catalyst, synthesis,  
 38 sensing, remediation, computing) with unique dynamic attrib-  
 39 utes including self-regulation, environmental responsiveness,  
 40 and sustainability.

#### 41 ASSOCIATED CONTENT

#### Supporting Information.

This material is available free of charge via the Internet at  
<http://pubs.acs.org>.

Materials and methods. Additional data S1-S6 (PDF)

#### AUTHOR INFORMATION

##### Corresponding Author

\***Huan-Hsuan. Hsu**- Department of Biomedical Engineering,  
 Tufts University, Medford, MA 02155, United States;  
<https://orcid.org/0000-0001-9455-6816>;

Email: [Huan-Hsuan.Hsu@tufts.edu](mailto:Huan-Hsuan.Hsu@tufts.edu).

\***Xiaocheng Jiang**- Department of Biomedical Engineering,  
 Tufts University, Medford, MA 02155, United States;  
<https://orcid.org/0000-0002-3455-3653>;

Email: [Xiaocheng.Jiang@tufts.edu](mailto:Xiaocheng.Jiang@tufts.edu).

##### Authors

**Pu Deng**- Department of Biomedical Engineering, Tufts Uni-  
 versity, Medford, MA 02155, United States;

Thomas Lord Department of Mechanical Engineering & Mate-  
 rials Science, Duke University, Durham, North Carolina  
 27705, USA; <https://orcid.org/0000-0001-5190-5350>

**Wei qin Sheng**- Department of Biomedical Engineering, Tufts  
 University, Medford, MA 02155, United States

School of Electronic Information, Hangzhou Dianzi Univer-  
 sity, Hangzhou 310018, People's Republic of China

**Andrew Xu**- Department of Biomedical Engineering, Tufts  
 University, Medford, MA 02155, United States

**Yixin Zhang**- Department of Biomedical Engineering, Tufts  
 University, Medford, MA 02155, United States

**Xiaochuan Dai**- Department of Biomedical Engineering, Tufts  
 University, Medford, MA 02155, United States

**Richard Vo**- Department of Biomedical Engineering, Tufts  
 University, Medford, MA 02155, United States

##### Author Contributions

‡ P. D., W. S., and A. X. contributed equally to this work

- 1  
2 **Notes**  
3 The authors declare no competing financial interest.  
4
- 5 **ACKNOWLEDGMENT**  
6 The authors gratefully acknowledge support from National Science  
7 Foundation (NSF DMR-1652095), Tufts Faculty Research Award  
8 and TUBERS Program.  
9
- 10 **REFERENCES**  
11 (1) Logan, B. E.; Rossi, R.; Ragab, A.; Saikaly, P. E. Electroactive  
12 Microorganisms in Bioelectrochemical Systems. *Nat. Rev. Microbiol.* **2019**,  
13 *17* (May), 307–319. <https://doi.org/10.1038/s41579-019-0173-x>.  
14 (2) Lovley, D. R. Electrically Conductive Pili: Biological Function and  
15 Potential Applications in Electronics. *Curr. Opin. Electrochem.* **2017**, *4* (1),  
16 190–198. <https://doi.org/10.1016/j.coelec.2017.08.015>.  
17 (3) Zhang, Y.; Hsu, L. H.-H.; Jiang, X. Living Electronics. *Nano Res.* **2019**,  
18 *12* (1), 22–26. <https://doi.org/10.1007/s12274-019-2570-x>.  
19 (4) Logan, B. E. Exoelectrogenic Bacteria That Power Microbial Fuel Cells.  
20 *Nat. Rev. Microbiol.* **2009**, *7* (5), 375–381.  
21 <https://doi.org/10.1038/nrmicro2113>.  
22 (5) Biological, G.; Lovley, D. R. E-Biologics: Fabrication of Sustainable  
23 Electronics with “Green” Biological Materials. *Am. Soc. Microbiol.* **2017**,  
24 *8* (3), 1–7.  
25 (6) Bonanni, P. S.; Bradley, D. F.; Schrott, G. D.; Busalmen, J. P. **92**  
26 Limitations for Current Production in *Geobacter Sulfurreducens* Biofilms. **93**  
27 *ChemSusChem* **2013**, *6* (4), 711–720. **94**  
28 <https://doi.org/10.1002/cssc.201200671>. **95**  
29 (7) Sun, D.; Chen, J.; Huang, H.; Liu, W.; Ye, Y.; Cheng, S. The Effect of **96**  
30 Biofilm Thickness on Electrochemical Activity of *Geobacter* **97**  
31 *Sulfurreducens*. *Int. J. Hydrogen Energy* **2016**, *41* (37), 16523–16528. **98**  
32 <https://doi.org/10.1016/j.ijhydene.2016.04.163>. **99**  
33 (8) Zhang, T.; Nie, H.; Bain, T. S.; Lu, H.; Cui, M.; Snoeyenbos-West, O. **100**  
34 L.; Franks, A. E.; Nevin, K. P.; Russell, T. P.; Lovley, D. R. Improved **101**  
35 Cathode Materials for Microbial Electrosynthesis. *Energy Environ. Sci.* **102**  
36 **2013**, *6* (1), 217–224. <https://doi.org/10.1039/c2ee23350a>. **103**  
37 (9) Xie, X.; Hu, L.; Pasta, M.; Wells, G. F.; Kong, D.; Criddle, C. S.; Cui, **104**  
38 Y. Three-Dimensional Carbon Nanotube-Textile Anode for High **105**  
39 Performance Microbial Fuel Cells. *Nano Lett.* **2011**, *11* (1), 291–296. **106**  
40 <https://doi.org/10.1021/nl103905t>. **107**  
41 (10) Wang, G.; Qian, F.; Saltikov, C. W.; Jiao, Y.; Li, Y. Microbial **108**  
42 Reduction of Graphene Oxide by *Shewanella*. *Nano Res.* **2011**, *4* (6), 563–**109**  
43 570. <https://doi.org/10.1007/s12274-011-0112-2>. **110**  
44 (11) Ding, C.; Liu, H.; Zhu, Y.; Wan, M.; Jiang, L. Control of Bacterial **111**  
45 Extracellular Electron Transfer by a Solid-State Mediator of Polyaniline **112**  
46 Nanowire Arrays. *Energy Environ. Sci.* **2012**, *5* (9), 8517–**113**  
47 8521. <https://doi.org/10.1039/c2ee22269h>. **114**  
48 (12) Hsu, L. (Huan-H.); Deng, P.; Zhang, Y.; Nguyen, H. N.; Jiang, X. **115**  
49 Nanostructured Interfaces for Probing and Facilitating Extracellular **116**  
50 Electron Transfer. *J. Mater. Chem. B* **2018**, *6* (17), 3051–**117**  
51 3058. <https://doi.org/10.1039/C8TB01598H>. **118**  
52 (13) Kornienko, N.; Zhang, J. Z.; Sakimoto, K. K.; Yang, P.; Reisner, E. **119**  
53 Interfacing Nature’s Catalytic Machinery with Synthetic Materials for **120**  
54 Semi-Artificial Photosynthesis. *Nat. Nanotechnol.* **2018**, *13* (10), 890–899. **121**  
55 <https://doi.org/10.1038/s41565-018-0251-7>. **122**  
56 (14) Zhang, P.; Liu, J.; Qu, Y.; Zhang, J.; Zhong, Y.; Feng, Y. Enhanced **123**  
57 Performance of Microbial Fuel Cell with a Bacteria/Multi-Walled Carbon **124**  
58 Nanotube Hybrid Biofilm. *J. Power Sources* **2017**, *361*, 318–325. **125**  
59 <https://doi.org/10.1016/j.jpowsour.2017.06.069>. **126**  
60 (15) Wimpenny, J.; Manz, W.; Szewzyk, U. Heterogeneity in Biofilms **127**  
61 **2000**, *24*, 661–671. **128**  
62 (16) Nudelman, F.; Sommerdijk, N. A. J. M. Biomineralization as an **129**  
63 Inspiration for Materials Chemistry. *Angew. Chemie - Int. Ed.* **2012**, *51* (27), 6582–6596. <https://doi.org/10.1002/anie.201106715>.  
64 (17) Narayanan, K. B.; Sakthivel, N. Biological Synthesis of Metal **65**  
66 Nanoparticles by Microbes. *Adv. Colloid Interface Sci.* **2010**, *156* (1–2), 1–  
67 13. <https://doi.org/10.1016/j.cis.2010.02.001>.  
68 (18) Nguyen, P. Q.; Courchesne, N. M. D.; Duraj-Thatte, A.; **69**  
70 Praveschotinunt, P.; Joshi, N. S. Engineered Living Materials: Prospects **71**  
72 and Challenges for Using Biological Systems to Direct the Assembly of **73**  
74 Smart Materials. *Adv. Mater.* **2018**, *30* (19), 1–34. **75**  
76 <https://doi.org/10.1002/adma.201704847>. **77**  
78 (19) Hsu, L. H.-H.; Zhang, Y.; Deng, P.; Dai, X.; Jiang, X. Biosynthetic **79**  
80 Electronic Interfaces for Bridging Microbial and Inorganic Electron **81**  
82 Transport. *Nano Lett.* **2019**, *19*, 8787–8792. **83**  
84 <https://doi.org/10.1021/acs.nanolett.9b03573>. **85**  
86 (20) Hsu, L.; Deng, P.; Zhang, Y.; Jiang, X. Core/Shell Bacterial Cables: A **87**  
88 One-Dimensional Platform for Probing Microbial Electron Transfer. *Nano* **89**  
89 *Let.* **2018**, *18* (7), 4606–4610. **90**  
91 <https://doi.org/10.1021/acs.nanolett.8b01908>. **92**  
92 (21) Valentin, T. M.; Landauer, A. K.; Morales, L. C.; DuBois, E. M.; **93**  
94 Shukla, S.; Liu, M.; Stephens Valentin, L. H.; Franck, C.; Chen, P.-Y.; **95**  
96 Wong, I. Y. Alginate-Graphene Oxide Hydrogels with Enhanced Ionic **97**  
98 Tunability and Chemomechanical Stability for Light-Directed 3D Printing. **99**  
99 *Carbon N. Y.* **2019**, *143*, 447–456. **100**  
100 <https://doi.org/https://doi.org/10.1016/j.carbon.2018.11.006>. **101**  
101 (22) Belchik, S. M.; Kennedy, D. W.; Dohnalkova, A. C.; Wang, Y.; Sevinc, **102**  
102 P. C.; Wu, H.; Lin, Y.; Lu, H. P.; Fredrickson, J. K.; Shi, L. Extracellular **103**  
103 Reduction of Hexavalent Chromium by Cytochromes MtrC and OmcA of **104**  
104 *Shewanella Oneidensis MR-1*. *Appl. Environ. Microbiol.* **2011**, *77* (12), **105**  
105 4035–4041. <https://doi.org/10.1128/AEM.02463-10>. **106**  
106 (23) Stolz, J. F.; Oremland, R. S. Bacterial Respiration of Arsenic and **107**  
107 Selenium. *FEMS Microbiol. Rev.* **1999**, *23* (5), 615–627. **108**  
108 <https://doi.org/10.1111/j.1574-6976.1999.tb00416.x>. **109**  
109 (24) Edberg, S. C.; Rice, E. W.; Karlin, R. J.; Allen, M. J. *Escherichia Coli*: **110**  
110 The Best Biological Drinking Water Indicator for Public Health Protection. **111**  
111 *J. Appl. Microbiol.* **2000**, *88* (S1), 106S–116S. **112**  
112 <https://doi.org/10.1111/j.1365-2672.2000.tb05338.x>. **113**  
113 (25) Logan, B. E.; Rabaey, K. Conversion of Wastes into Bioelectricity and **114**  
114 Chemicals by Using Microbial Electrochemical Technologies. *Science (80-* **115**  
115 *).* **2012**, *337* (6095), 686–690. <https://doi.org/10.1126/science.1217412>. **116**  
116 (26) Logan, B. E.; Hamelers, B.; Rozendal, R.; Schröder, U.; Keller, J.; **117**  
117 Freguia, S.; Aelterman, P.; Verstraete, W.; Rabaey, K. Microbial Fuel Cells: **118**  
118 Methodology and Technology. *Environ. Sci. Technol.* **2006**, *40* (17), 5181–**119**  
119 5192. <https://doi.org/10.1021/es0605016>. **120**  
120 (27) Yan, F. F.; Wu, C.; Cheng, Y. Y.; He, Y. R.; Li, W. W.; Yu, H. Q. **121**  
121 Carbon Nanotubes Promote Cr(VI) Reduction by Alginate-Immobilized **122**  
122 *Shewanella Oneidensis MR-1*. *Biochem. Eng. J.* **2013**, *77*, 183–189. **123**  
123 <https://doi.org/10.1016/j.bej.2013.06.009>. **124**  
124 (28) Choe, G.; Oh, S.; Seok, J. M.; Park, S. A.; Lee, J. Y. Graphene **125**  
125 Oxide/Alginate Composites as Novel Biointerfaces for Three-Dimensional **126**  
126 Mesenchymal Stem Cell Printing and Bone Regeneration Applications. **127**  
127 *Nanoscale* **2019**, *11* (48), 23275–23285. **128**  
128 <https://doi.org/10.1039/c9nr07643c>. **129**  
129 (29) Lee, K. Y.; Mooney, D. J. Alginate: Properties and Biomedical **130**  
130 Applications. *Prog. Polym. Sci.* **2012**, *37* (1), 106–126. **131**  
131 <https://doi.org/10.1016/j.progpolymsci.2011.06.003>. **132**  
132 (30) Tanaka, H.; Matsumura, M.; Veliky, I. A. Diffusion Characteristics of **133**  
133 Substrates in Ca-alginate Gel Beads. *Biotechnol. Bioeng.* **1984**, *26* (1), 53–  
134 58. <https://doi.org/10.1002/bit.260260111>. **135**

1	50
2	51
3	52
4	53
5	54
6	55
7	56
8	57
9	58
10	59
11	60
12	61
13	62
14	63
15	64
16	65
17	66
18	67
19	68
20	69
21	70
22	71
23	72
24	73
25	74
26	75
27	76
28	77
29	78
30	79
31	80
32	81
33	82
34	83
35	84
36	85
37	86
38	87
39	88
40	89
41	90
42	91
43	92
44	93
45	94
46	95
47	96
48	97
49	98

---

For table of contents only

