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Hydrothermal synthesis of carbon nanotube-titania composites for enhanced photocatalytic performance

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Nanosized, well-dispersed titania particles were synthesized via a hydrothermal method using multiwalled carbon nanotubes (MWCNTs) as structural modifiers during the nucleation process to decrease aggregation. Synthesized TiO₂/MWCNT composites containing different amounts of MWCNTs were characterized using N₂ physisorption, XRD, spectroscopic techniques (Raman, UV-visible, and X-ray photoelectron), and electron microscopy to illuminate the morphology, crystal structure, and surface chemistry of the composites. Photocatalytic performance was evaluated by measuring the degradation of acetaldehyde in a batch reactor under UV illumination. Average rate constants decrease in the following order: TiO₂/MWCNT-1% > TiO₂ > TiO₂/MWCNT-5%. Addition of MWCNTs beyond the optimum loading ratio of 1:100 (MWCNT:TiO₂) diminishes the effectiveness of the photocatalyst and the synergistic effect between MWCNTs and TiO₂. The primary mechanism for photocatalytic activity enhancement in TiO₂/MWCNT-1% is thought to be due to increased porosity, hydroxyl enrichment on the surface, and high dispersion of TiO₂ particles.

Introduction

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Photocatalytic oxidation (PCO) using semiconductors as catalysts has become an appealing approach as a green technology for environmental remediation since the discovery of UV-assisted splitting of water on TiO_2 by Fujishima and Honda [1]. Air contaminants such as SO_∞ NO $_\infty$ halogenated organics, and volatile organic compounds (VOCs) are suitable candidates for PCO [2, 3, 4, 5, 6, 7, 8, 9, 10]. TiO_2 has been identified as the most promising photocatalytic material based on its strong oxidizing power, chemical stability, relatively low price, and biological benignity [11, 12]. However, photocatalytic performance of conventional TiO_2 is limited by its wide band gap (3.2 eV for the anatase crystal phase [13]) and fast electron-hole recombination. These properties restrict the photon absorption range to ultraviolet wavelengths (only 8.3% of total solar irradiance [14]) and limit the quantum efficiency [15, 16].

In an effort to overcome these problems and increase the photocatalytic activity of TiO_2 , researchers have explored a number of modification approaches [17, 18, 19, 20] and

recently turned to carbon nanotubes (CNTs) and other carbon nanomaterials [21, 22, 23, 24]. CNTs, which are either singlewalled (SWCNTs) or multiwalled (MWCNTs), possess unique properties that make them the ideal supports for catalysts such as large surface area (>200 m²/g), high thermal and chemical stability, as well as excellent electrical and mechanical properties. In particular, CNTs have a large electron-storage capacity (one electron for every 32 carbon atoms) and thus can serve as an electron sink; in addition, the highly conductive nature of CNTs promotes charge separation via the creation of heterojunctions at the CNT-TiO₂ interface [25]. A number of studies [21, 22, 23, 24, 25, 26, 27, 28, 29, 30, 31] have shown the unique electronic properties of CNTs can be exploited to induce beneficial charge transfer properties and synergistic effects between carbon phases and TiO2; thus coupling CNTs to TiO₂ can improve the photocatalytic performance of TiO₂. However, the exact role of CNTs in TiO2/CNT composites is still poorly understood especially as it can be convoluted by factors such as synthesis methods, CNT morphologies, and

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weight ratio of CNTs to TiO₂. Further insight into their syntheses and mechanisms is clearly required.

The conventional method for synthesizing TiO2/CNT include sol-gel and hydrothermal processes [32]. Sol-gel synthesis utilizes inorganic metal salts or metal organic compounds to produce a sol, which transitions from a liquid to a gel upon polymerization of the precursor and loss of solvent. The hydrothermal method [33] used in this study involves heating a solution containing the titanium precursor to temperatures above the boiling point of the solvent, resulting in significantly increased pressures within the autoclave reactor. Synthesis parameters such as heating time in the autoclave, temperature, and pH affect the properties of resulting TiO2 and offer a reliable path for improved control of the composite properties. Low temperature and relatively short reaction time proved best for synthesis of small TiO2 particles, whereas mild acidity led to control of the crystal phase and yielded pure anatase [33, 34]. The hydrothermal method described earlier has been used by Zhang et al. [35] for creating TiO2/CNT composites, whereas Jitianu et al. [36] showed TiO2 decorations on MWCNTs were possible. Silva et al. [37] also showed that microwave-assisted hydrothermal process can be used for high mass loading of metal oxide and CNT composites. Hydrothermal synthesis of TiO2/MWCNT composites was explored in this work due to the possibility of controlling the TiO₂ phase by controlling pH during the TiO2 nucleation step in the hydrothermal aging process. To the best of our knowledge, no studies have been reported on the degradation of acetaldehyde, a common indoor pollutant [38, 39], using a hydrothermally prepared mesoporous CNT/TiO2 photocatalyst. Moreover, whereas many studies involving TiO2/CNT composites have focused on liquid-phase degradation of dyes, few have investigated gas-phase VOC degradation [40, 41].

In this work, a hydrothermal synthesis method has been used to synthesize TiO₂/MWCNT catalyst composites for efficient degradation of acetaldehyde under UV illumination. The hydrothermal method shows promise in controlling the anatase/rutile ratio, as well as producing diameter-controlled crystalline TiO₂ particles on a large scale [42]. Catalysts were synthesized with nominal mass ratios of 1:100 (TiO₂/MWCNT-1%), 1:20 (TiO₂/MWCNT-5%), and pure TiO₂ (100% TiO₂) as a reference. Performance of the composites was tested in the photocatalytic degradation of acetaldehyde in a batch reactor.

Results and discussion

Characterization of photocatalysts

XRD was used to characterize the phase composition of TiO_2 and $TiO_2/MWCNT$ composites, and their diffraction patterns are presented in Fig. 1. The anatase peaks at 2θ values of 25.40° ,

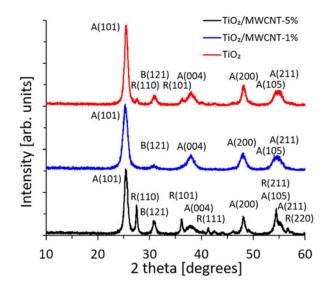


Figure 1: XRD patterns of TiO₂ and TiO₂–MWCNT nanocomposites. Peak labels have been included to correlate peaks to crystal phase [anatase (A), rutile (R), or brookite (B)].

37.9°, 48.0°, 53.9°, and 55.1° can be indexed to (101), (004), (200), (105), and (211) crystal planes, respectively. Diffraction peaks for rutile at 27.4°, 36.2°, 41.3°, 54.4°, and 56.7° are present, which are indexed to (110), (101), (111), (211), and (220) crystal planes. The primary diffraction peak for brookite at 30.9°, indexed to (121), is also present. These peak assignments are also in agreement with prior studies [43, 44, 45, 46]. Comparing the reflection peaks observed in TiO₂ with the TiO₂/MWCNT composites, it can be concluded the dominant phase in all photocatalysts is anatase, although the TiO₂/ MWCNT-5% composite appears to have a substantial fraction of rutile. A summary of the complete phase composition of the photocatalysts is presented in Table SI. The height ratio of the dominant anatase peak at 25.2° to the dominant rutile peak at 27.4° ($I_{\text{Ana}}/I_{\text{Rut}}$) can be used to determine the weight fraction of anatase and rutile crystalline phases of TiO2 from Eq. (1) as demonstrated by Spurr and Myers [47]; the calculated mass ratios of anatase to rutile in TiO2, TiO2/MWCNT-1%, and TiO₂/MWCNT-5% are 66:34, 98:2, and 92:8, respectively. It is worth mentioning the characteristic peak of MWCNTs (002) located at $2\theta = 25^{\circ}$ is not observed for the $TiO_2/MWCNT$ samples. This is most likely due to the overlap with the strong anatase (101) reflections.

$$f = \frac{1}{1 + 1.26 \left(\frac{I_{\rm R}}{I_{\rm A}}\right)} \quad . \tag{1}$$

Average TiO₂ particle size for the samples (Table I) was calculated from the full width at half maximum (FWHM) of the principal anatase (101) peak at 25.2° using the Williamson–Hall method. Average TiO₂ particle sizes for TiO₂, TiO₂/

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MWCNT-1%, and TiO₂/MWCNT-5% composites are 6.1, 5.8, and 7.2 nm, respectively. The high uniformity in TiO₂ particle sizes for the catalyst composites indicates the addition of MWCNTs during synthesis did not affect the size. Analysis of the $E_{\rm g}$ band (\approx 142 cm⁻¹) in the Raman spectra (Fig. S1) confirms the insignificant differences in particle size, as size discrepancies would manifest as an increase in peak frequency [48]. Scanning electron micrograph (SEM) images of the synthesized photocatalysts presented in Fig. 2 further illustrate the similarity of the bulk powder morphologies. Using energy-dispersive X-ray spectroscopy (EDX), carbon amounts were estimated at 1.4 and 6.6 wt% for TiO₂/MWCNT-1% and TiO₂/MWCNT-5% catalysts, respectively; these values indicate good agreement with the nominal values of 99 and 95 wt% TiO₂ in

TABLE I: Summary of catalyst properties.

Catalyst characterization				
	TiO ₂	TiO ₂ /MWCNT-1%	TiO ₂ /MWCNT-5%	
Surface area (m ² /g) ^a	74.6	141.6	68.2	
Total pore volume (cm ³ /g) ^a	0.14	0.30	0.16	
Average pore diameter (nm) ^a	7.60	8.36	9.42	
TiO ₂ average diameter (nm) ^b	6.1	5.8	7.2	
Anatase content (wt%) ^b	92	100	67	
Carbon content (wt%) ^c	N/A	1.4	6.6	

^aNitrogen physisorption.

the samples. EDX data shown in Fig. 3 reveal TiO₂/MWCNT-5% contains pockets of carbon in different regions exceeding 10 wt%, which may indicate CNT clustering during the TiO₂ nucleation phase. Intimate contact between MWCNTs and TiO₂ is apparent in the transmission electron microscopic (TEM) images for TiO₂/MWCNT-5% in Fig. 4, even after 10 min of sonication during the sample preparation process for TEM characterization; this provides evidence of strong interactions between MWCNTs and TiO₂ nanoparticles, and high dispersion, especially for TiO₂/MWCNT-1%. Particle sizes estimated from the TEM images for the samples are in the range of 3–8 nm. The slight discrepancy in particle size obtained from TEM and XRD is consistent with overestimation of the size of crystallite particles by the Scherrer equation [49, 50].

Brunauer–Emmet–Teller (BET) surface areas of pure TiO_2 , $TiO_2/MWCNT-1\%$, and $TiO_2/MWCNT-5\%$ were determined to be 74.6, 141.6, and 68.2 m²/g, respectively. The adsorption–desorption isotherms are presented in Fig. S2. The dramatic difference in surface area clearly indicates an effect of MWCNTs on the morphology of the photocatalysts, which could stem from the presence of MWCNTs during the nucleation of TiO_2 nanoparticles as part of the hydrothermal process. A small amount of MWCNTs (\sim 1 wt%) effectively increased the surface area of resulting composites, whereas a further increase in the amount of MWCNTs (\sim 5 wt%) resulted in particle agglomeration, reducing the overall surface

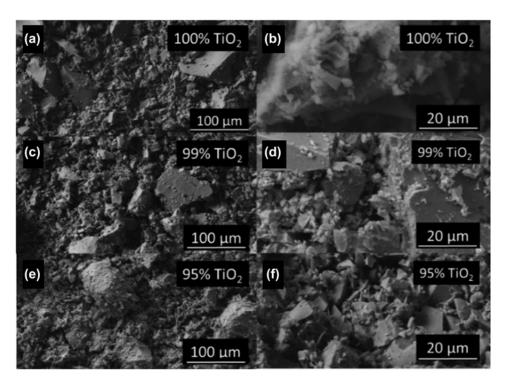


Figure 2: Low- and high-magnification SEM images of photocatalysts: TiO₂ (a, b), TiO₂/MWCNT-1% (c, d), and TiO₂/MWCNT-5% (e, f).

bXRD.

cEDX.



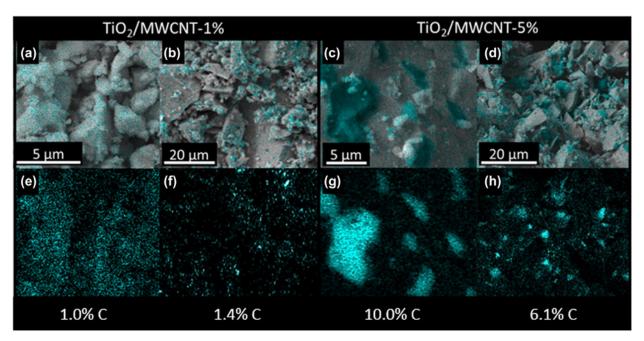


Figure 3: Carbon dispersion within photocatalysts. SEM images with EDS mapping of elemental carbon overlayed (a–d). EDS mapping of elemental carbon illustrating carbon dispersion.

area of the catalyst composite. Total pore volumes and average pore sizes determined by the Barrett-Joyner-Halenda (BJH) method for pure TiO₂, TiO₂/MWCNT-1%, and TiO₂/MWCNT-5% are summarized in Table I. The existence of mesopores in the catalyst composites is evidenced by pore sizes in the order 7.6, 8.4, and 9.4 nm for TiO₂, TiO₂/MWCNT-1%, and TiO₂/MWCNT-5%, respectively. Significant increase in porosity is observed in the TiO₂/MWCNT-1% composite, evidenced by substantially greater pore volume despite comparable pore size. Use of a hydrothermal method with a low weight fraction of MWCNTs yields TiO₂/MWCNT composites characterized by a combination of large specific surface area and pore size distributions in the mesopore range—features beneficial in catalysis.

X-ray photoelectron spectroscopic (XPS) characterization was performed to elucidate the surface composition of the photocatalysts. Figure 5 shows high-resolution XPS spectra for the O 1s region of photocatalysts fitted using two asymmetric peak components. The peak component at binding energy of ~531 eV is attributed to the oxygen bonded to Ti (Ti-O) in TiO₂, whereas the component at ~532 eV is related to the hydroxyl group [51]. Results of deconvolution of O 1s peak for the photocatalysts are summarized in Table II. Percentage concentrations of the two oxygen species identified changes with the amount of MWCNTs used during synthesis. Hydroxyl content in the photocatalysts decreases in the following order: TiO₂/MWCNT-1% > TiO₂/MWCNT-5% > TiO₂. This result suggests the presence of small amounts of MWCNTs (~1 wt%) enhances chemisorption of water on the surface of TiO₂—

a feature hardly observed for composites with 5 wt%. Prior studies have shown chemisorption of water on TiO₂ composites is enhanced with the addition of CNTs [51, 52]. Since XPS analysis occurs under ultrahigh vacuum, the observed hydroxyl species is a result of chemisorption, not physisorption. XPS analysis of the catalyst composites reveals a substantially higher percentage of chemisorbed hydroxyl groups on the surface of the TiO₂/MWCNT-1%. Hydroxyl enrichment on the surface of TiO₂ promotes photocatalytic activity due to an increase in the generation of •OH radicals from the reaction between photogenerated holes and chemisorbed hydroxyl ions (or moisture) on the surface of photocatalyst. Therefore, synthesis approaches that yield high amounts of hydroxyl groups on the surface benefit photocatalytic reactions.

Photocatalytic activity

The activity of photocatalysts was measured by evaluating the rate of acetaldehyde degradation in the gas phase under UV light illumination. Figure 6 shows data for degradation of acetaldehyde over pure ${\rm TiO_2}$, ${\rm TiO_2}/{\rm MWCNT}$ -1% and ${\rm TiO_2}/{\rm MWCNT}$ -5%. It is clear from the results that ${\rm TiO_2}/{\rm MWCNT}$ -1% shows the highest photocatalytic performance because the rate of decrease of relative concentration (C/C_o) is the fastest among the photocatalysts [Fig. 6(a)]. Photocatalytic degradation of acetaldehyde is assumed to be a pseudo–first-order reaction [53] with the following simplified Langmuir–Hinshelwood kinetic model when initial concentration (C_o) is negligible:

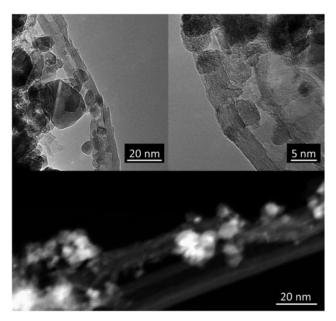


Figure 4: TEM images of TiO₂ particles on MWCNTs in the TiO₂/MWCNT-5% sample.

$$\ln\left(\frac{C}{C_0}\right) = -k_0 t \quad ,$$
(2)

where $k_{\rm o}$ is the apparent first-order rate constant. From the kinetic plots in Fig. 6(b), the TiO₂/MWCNT-1% shows the highest slope and, consequently, the highest average rate constant [Fig. 6(c)]. By contrast, composites prepared without MWCNTs (pure TiO₂), or with relatively high amounts of MWCNTs (TiO₂/MWCNT-5%), exhibit inferior photocatalytic performance. The respective shades around the plots in Fig. 6 show the range within one standard deviation for all data collected at each time interval. Projection lines intercept the *y*-axis after time zero due to desorption of acetaldehyde when the UV lights were switched on.

The average rate constant for acetaldehyde degradation with TiO_2 is $8.23 \times 10^{-2} \text{ min}^{-1}$. In particular, degradation reactions with TiO2/MWCNT-1% resulted in a photocatalytic rate constant of $15.0 \times 10^{-2} \text{ min}^{-1}$, which is higher than that of a pure TiO2 photocatalyst by almost a factor of two. By comparison, Sano et al. [54] saw a 60% increase in acetaldehyde degradation rate using a platinum-doped TiO2 catalyst as compared with P25 in humid air, whereas Vijayan et al. [20] achieved a 40% increase in the reaction rate for acetaldehyde PCO under UV illumination with a titania nanotube/SWCNT composite. It is apparent introducing a small amount of MWCNTs into a TiO2 matrix enhances the photocatalytic activity of the resulting TiO₂-based composites. Conversely, the rate constant obtained for the degradation reaction with TiO2/MWCNT-5% is $6.50 \times 10^{-2} \text{ min}^{-1}$, which is surprisingly lower than that of pure TiO2; this indicates a further addition of MWCNTs, beyond the optimum loading ratio of 1:100 (MWCNTS:TiO₂), diminishes the effectiveness of the photocatalyst.

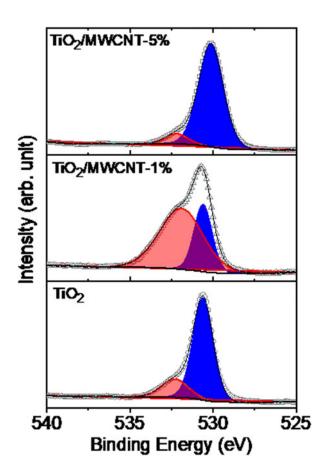


Figure 5: XPS high-resolution O 1s spectra of photocatalysts fitted using two asymmetric peak components.

As mentioned earlier, the $TiO_2/MWCNT-5\%$ catalyst is characterized by a lower percentage of anatase (\sim 67 wt%). The observed difference in photocatalytic performance is not

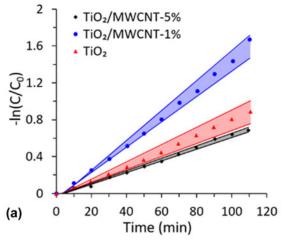


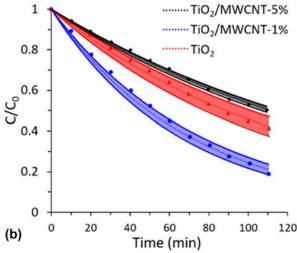
TABLE II: Summary of results of deconvolution of high-resolution XPS O 1s spectra of TiO_2 and TiO_2 /MWCNT composites using peak components with asymmetric line shapes.

Materials	O 1s (Ti-O)	O 1s (OH)
TiO ₂		
Binding energy (eV)	530.1	532.2
Peak area	38941.7	3402.4
% Concentration	91.97	8.03
TiO ₂ /CNT-1%		
Binding energy (eV)	530.6	531.9
Peak area	10104.5	22672.0
% Concentration	30.84	69.16
TiO ₂ /CNT-5%		
Binding energy (eV)	530.6	532.3
Peak area	32081.9	7597.3
% Concentration	80.86	19.14

attributed to the composition of the crystal phases because the ideal ratio of anatase to rutile for maximal PCO rates has been shown to be between 40 and 80% anatase [55]. Electronic structure calculations for mixed rutile–anatase systems have revealed band alignment of rutile lies higher than that of anatase, resulting in hole accumulation in the valence band of rutile [56]. This conclusion supports the idea that higher rutile amounts than that in P25 may be beneficial to photocatalytic reactions that utilize holes generated in the valence band.

The TiO2/MWCNT-1% composite is able to achieve 80% degradation of acetaldehyde after 110 min, whereas pristine TiO₂ achieves less than 60% degradation for the same illumination time. The primary mechanism for photocatalytic activity enhancement in TiO2/MWCNT-1% is thought to be the result of structural effects from the addition of small amounts of MWCNTs. When higher amounts of MWCNTs are present, CNT agglomeration within the bulk catalyst occurs, resulting in TiO₂ nucleation around CNT bundles. This phenomenon is illustrated in Fig. 7. Whereas the average TiO2 particle sizes are comparable, stark differences in surface area and pore volume suggest dispersion is affected by the presence of CNTs during TiO₂ particle nucleation. Without the presence of CNTs, TiO₂ particles experience some agglomeration during nucleation. Adding a small amount of MWCNTs (~1%) increased dispersion while providing nucleation sites for TiO2 particle formation. Further addition of MWCNTs results in CNT clustering in the aqueous solution, due to their high hydrophobicity despite the oxidation of the MWCNTs. MWCNT clustering can be observed in both EDX (Fig. 3) and Raman spectroscopic results (Fig. S1). Raman spectra for the TiO₂/ MWCNT-5% sample show substantial differences in intensity of the characteristic peaks of MWCNTs (i.e., the D-band ~1350 cm⁻¹ and G-band ~1590 cm⁻¹) between the two spectra obtained from two different spots. The spectrum with carbon peaks of higher intensity exemplifies the agglomeration of CNTs in the scanned region.





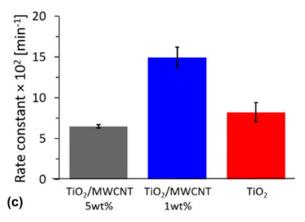


Figure 6: (a) Degradation curves of photocatalyst performance projection. Center lines indicate the first-order rate equation derived from the average degradation data for each catalyst. (b) Photocatalytic degradation of acetal-dehyde as a function of UV illumination time. (c) Comparison of the average rate constants for the photocatalytic degradation of acetaldehyde using TiO₂/MWCNT-1%, and TiO₂/MWCNT-5%.

MWCNT clustering results in agglomeration of TiO₂ particles around the MWCNT bundles in addition to the typical nucleation and aggregation experienced in pristine

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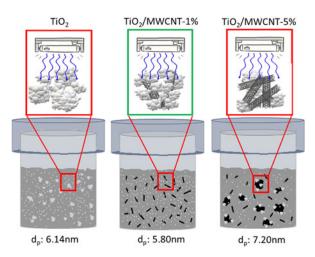


Figure 7: Proposed mechanism for the role of CNTs in TiO₂ nucleation and dispersion during hydrothermal synthesis of TiO₂/MWCNTs.

 ${
m TiO_2}$ solution. As a consequence, the surface area and total pore volume of the photocatalyst decrease, resulting in a lower photocatalytic performance. In addition, CNT light absorption may begin to hinder the effectiveness of ${
m TiO_2}$ as the active photocatalytic material [57]. UV-vis diffuse reflectance spectroscopy (DRS) data provided in Fig. S3 confirm the photocatalytic activity enhancement in the ${
m TiO_2/MWCNT-1\%}$ catalyst is not due to band gap reduction as it appears to have blue-shifted subtly compared with the ${
m TiO_2}$ catalyst. These results support the hypothesis of photocatalytic enhancement from MWCNT addition due to physical effects.

Conclusions

CNT/TiO₂ catalysts of different CNT:TiO₂ ratios were synthesized using a hydrothermal method. The addition of only 1 wt% MWCNTs resulted in nearly double the degradation rate of acetaldehyde vapor. The observed catalytic enhancement is attributed to morphological differences caused by the presence of MWCNTs during TiO₂ particle nucleation and change in surface chemistry of the photocatalysts. The presence of small amounts of MWCNTs (~1 wt%) enhances chemisorption of water on the surface of TiO₂. Further addition of MWCNTs corresponding to 5 wt% resulted in lower photocatalytic performance, which is attributed to the agglomeration of CNTs during the hydrothermal process and possible reduction, as well as interference in the absorption of light by TiO₂.

Experimental

Chemicals

MWCNTs (purity >95%, outer diameter 20–30 nm, and length 10–30 μ m) were obtained from Chengdu Organic Chemicals

Co., Ltd., Hydrogen peroxide (30%) was obtained from Fisher Scientific. Titanium tetrachloride (TiCl₄, purity >99.0%) was obtained from Strem Chemicals. Sodium hydroxide (NaOH, purity >99%) and acetaldehyde (C₂H₄O, purity >99%) were obtained from Sigma-Aldrich. Dry air was obtained from Matheson.

CNT oxidation

MWCNTs were photo-oxidized in 30 wt% $\rm H_2O_2$ via UV illumination (White-Rodger Comfort Plus UV200, 2 Philips TUV PL-L 60 W bulbs). UV-assisted $\rm H_2O_2$ oxidation of MWCNTs has been shown to be effective and less aggressive in comparison to conventional acid oxidation using HNO₃ [58, 59]. In a typical batch, 0.5 g of MWCNTs were added to 200 mL of $\rm H_2O_2$, which was maintained at 60 °C under UV illumination and vigorous stirring. The MWCNTs were oxidized for 72 h, with 50 mL of $\rm H_2O_2$ being added at 24 and 48 h to sustain the process. After 72 h, the oxidized MWCNTs were removed from solution via vacuum filtration and rinsed with distilled water. Oxidized MWCNTs were obtained after the samples were vacuum-dried overnight at 60 °C.

Catalyst synthesis

Oxidized MWCNTs were dispersed in 50 mL H₂O via sonication at room temperature for 3 h. Following sonication, TiCl₄ amounts corresponding to TiO₂:MWCNT ratios of 99:1 and 95:5 were added dropwise at a rate of 1 mL/min, to the dispersed MWCNTs at ~5 °C under moderate stirring (0.88 mol/L Ti). After addition of the TiCl₄ and resulting exothermic reaction, the solution was allowed to cool from ~25 to 10 °C under continuous stirring. Once the solution reached 10 °C, 35 mL of 5 M NaOH were added to the solution to obtain a pH of 3-5. The addition of the strong base led to the formation of abundant NaCl, causing the solution to become highly viscous. The stir rate was increased gradually from 300 to 800 rpm as the viscosity increased. When the solution reached a pH range of 3-5, water was added under constant stirring to produce a solution of 0.44 mol/L Ti. The solution was then put in a steel autoclave and heated in the oven for 3 h at 150 °C, removed, and allowed to cool for 3 h. The TiO₂/MWCNT composite was dried by air continuously flowing over it for 24 h followed by 3 h in the oven at 80 °C. Once dried, TiO2/MWCNT was crushed into a fine powder and baked at 250 °C for 3 h. To remove NaCl, the composite was rinsed via vacuum filtration with 250 mL of deionized water and allowed to dry for 12 h. Finally, the composite was calcined at 450 °C for 3 h (a temperature low enough to prevent TiO₂ sintering) [60], cooled, and stored as a powder. Before evaluation of the catalyst performance, the powders were sieved to <125 µm to reduce particle agglomeration and grain size distribution.



Photocatalysts were synthesized with nominal mass ratios of 1:100 (TiO₂/MWCNT-1%), 1:20 (TiO₂/MWCNT-5%), and pure TiO₂ (100% TiO₂) as a reference.

Characterization

XRD patterns were obtained using a Rigaku MiniFlex II diffractometer with a Cu Ka radiation source. Two theta scanning angles from 10° to 80° degrees were taken with a step size of 0.02° and scan speed of 2.0°/min for all photocatalysts. Surface area and pore volume were derived from N2 adsorption-desorption measurements at -196 °C. The BET method [61] was applied to the adsorption isotherm in the relative pressure range of 0.02-0.35 to calculate the specific surface area of the synthesized photocatalysts. Pore-size distribution was obtained from the desorption branch at a relative pressure of 0.99 of the isotherm using the BJH model [62]. SEM were obtained using a Hitachi S5200, operated at 10 kV and equipped with EDX. TEM images were obtained with an FEI Tecnai F20 XT equipped with EDX at 120 kV. A small amount of photocatalyst powder was dispersed in ethanol by sonication for 10 min, and a drop of the solution was deposited onto a carbon TEM grid. Raman spectroscopy was obtained from a Renishaw Raman spectrometer using a CW laser with a wavelength of 514 nm. Raman spectra were collected at different locations of the sample. The TiO₂ and TiO₂/MWCNT samples were analyzed by UV-visible diffuse reflectance spectroscopy (UV-vis DRS) using a Shimadzu UV-2600 spectrometer with BaSO₄ as the background material. XPS measurements were conducted on a Kratos Ultra XPS system with a standard Mg K_{α} source (hv = 1253.6 eV). A charge neutralizer was employed to minimize the effect of charging. High-resolution spectra were acquired at 20 eV pass energy with 0.1 eV steps. XPS spectra were analyzed using CasaXPS software. Peak components in the high-resolution spectra were fit with a combination of Gaussian and Lorentzian peak shapes.

Photocatalytic performance

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In a standard catalyst evaluation, 0.100 ± 0.001 g of sieved photocatalyst powder was loaded into a borosilicate sample dish, which was then placed in a 750-mL borosilicate batch reactor. The reactor was purged using UHP air (low CO₂ and H₂O content) at 6 liters per minute (LPM) for 5 min. Thereafter, acetaldehyde vapor corresponding to 1000 ppmv was injected into the reactor and allowed to equilibrate in darkness for 90 min. The ultraviolet illumination system—consisting of two 25 W Sylvania 21703 (356 nm) bulbs and two 13 W ReptiSun 10.0 UV/B bulbs (emission spectrum displayed in Fig. S1)—was switched on to initiate the reaction for a duration of 120 min. One-mL gas samples were taken from the reactor at \sim 10 min intervals for analysis. The decrease in

acetaldehyde concentration and increase in CO2 concentration were monitored using an SRI 310C gas chromatograph equipped with a Restek silica gel column (8046-895, 6 ft length, 2 mm ID), flame ionization detector (FID), and on-column injection. Detected peaks were integrated using PeakSimple Chromatography Integration software. The relative concentration is expressed as C/C_0 , where C is the concentration of acetaldehyde at time t and C_0 is the initial acetaldehyde concentration. The photocatalytic experiment was conducted at room temperature (~25 °C) with an indoor relative humidity of ~50%. The photocatalytic performance of three separate 0.100 g samples from each composite ratio were tested to ensure statistical significance of the results. Average catalyst performance was calculated using the average concentration of three degradation experiments for each catalyst. First-order rate assumptions were applied in analysis of the degradation data.

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Supplementary material

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