

Evolution and Reversible Polarity of Multilayering at the Ionic Liquid/Water Interface

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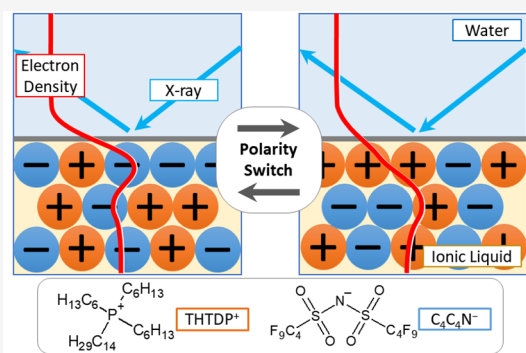


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ABSTRACT: Highly correlated positioning of ions underlies Coulomb interactions between ions and electrified interfaces within dense ionic fluids such as biological cells and ionic liquids. Recent work has shown that highly correlated ionic systems behave differently than dilute electrolyte solutions, and interest is focused upon characterizing the electrical and structural properties of the dense electrical double layers (EDLs) formed at internal interfaces. It has been a challenge for experiments to characterize the progressive development of the EDL on the nanoscale as the interfacial electric potential is varied over a range of positive and negative values. Here we address this challenge by measuring X-ray reflectivity from the interface between an ionic liquid (IL) and a dilute aqueous electrolyte solution over a range of interfacial potentials from -450 to 350 mV. The growth of alternately charged cation-rich and anion-rich layers was observed along with a polarity reversal of the layers as the potential changed sign. These data show that the structural development of an ionic multilayer-like EDL with increasing potential is similar to that suggested by phenomenological theories and MD simulations, although our data also reveal that the excess charge beyond the first ionic layer decays more rapidly than predicted.



INTRODUCTION

Ions are densely packed in ionic liquids (ILs)—which are solvent free—and other systems such as concentrated polyelectrolyte solutions and biological cells. The positions of these ions are highly correlated as the result of excluded volume interactions and strong electrostatic interactions between neighboring ions.^{1,2} As a result of these strong correlations, the electrical double layer (EDL) structure of ionic liquids (ILs) differs from conventional dilute electrolyte solutions.^{3,4} The structure of the EDL underlies electrostatic interactions between ions and electrified interfaces such as electrodes in electrochemical systems.

Molecular dynamics simulations have observed the formation of EDLs consisting of ionic multilayers at the interface of ILs with a solid electrode.^{5–12} These studies revealed a gradual change from mixed cation–anion multilayers at an uncharged electrode to alternating cation-rich and anion-rich layers upon charging the electrode.^{13,14} Once the electrode is sufficiently charged, the first layer of ions adjacent to it overscreens the electrode; that is, the charge in this first layer exceeds the charge on the electrode. The second layer is enriched with co-ions and overscreens the first layer but to a lesser extent. The sign of the electrode charge determines the polarity of the alternately charged ionic multilayers. Over-screening is a direct consequence of electrostatically induced spatial correlations¹⁵ and does not appear in conventional

theories of dilute electrolyte solutions, such as the Gouy–Chapman theory.^{16,17} Furthermore, MD^{13,14} and theoretical^{18,19} studies predict that increasing the charge on the electrode produces crowding of ions, which occurs when the electrode charge is large enough that a complete, single layer of counterions is insufficient to screen it and a second counterion-rich layer is required.

Experimental techniques that have been used to study the distribution of ions in the EDL structure of ILs include X-ray reflectivity (XR),^{20–28} neutron reflectivity,^{29–32} atomic force microscopy,^{33–41} and surface force measurements.^{42–46} However, results from experiments vary in several important aspects. For example, the measured thickness of the EDL ranges from a Helmholtz-like single ion dimension^{32,47} in some experiments to several²⁰ and even tens⁴⁶ of ions thick in others. XR studies that measure the EDL between ILs and a variety of solid electrode materials, such as Au,²³ Si,²⁷ epitaxial graphene on SiC,^{24,28} and boron-doped diamond,²⁶ have exhibited different interfacial structures. A thick interfacial layer of

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several nm was observed at positive potentials for an IL in contact with a Si electrode,²⁷ but it was not observed with a boron-doped diamond electrode.²⁶ Instead, as the electric potential was varied from +1.5 to −2.5 V, the surface charge density decreased by only 20% of a monolayer equivalent; the polarity of the ionic multilayers did not change, in spite of reversing the potential. Two studies have suggested a change in polarity upon reversing the potential,^{23,24} but the systematic growth and reversal of polarity as the interfacial potential is scanned from negative to positive potentials has not been observed experimentally. Here, we present such observations by utilizing X-ray reflectivity measurements to characterize the development of alternately charged ionic multilayers on the molecular scale.

In contrast to these studies of the IL/solid electrode interface, we study the electrochemical liquid/liquid interface between IL and water (W)⁴⁸ using XR methods similar to those used earlier to study the electrochemical liquid/liquid interface between dilute organic and aqueous electrolyte solutions.^{49–52} The soft interface between an IL and an aqueous electrolyte solution may provide a useful contrast to the study of IL against hard electrode surfaces for several reasons. First, the effect of structural defects present on most solid surfaces is avoided. Second, electrostriction effects, in which ions pack closer to a solid charged electrode with an enhanced surface density as a result of an increased electrostatic attraction to the surface,^{13,14} should be reduced at a soft interface which can be distorted by the ions. Third, the geometric boundary condition imposed by a solid substrate induces multilayering. Many liquids, with or without ions, form molecular multilayers at the liquid/solid interface; even pure water does this,⁵³ although it does not form multilayers at the soft water/vapor interface.⁵⁴ Nevertheless, the formation of cation–anion mixed multilayers at the soft IL/vapor interface has been demonstrated by us,^{25,55,56} and others.^{57,58} For example, our earlier XR measurements of the EDL structure at the IL/W interface for two potentials that straddle the potential of zero charge (PZC) demonstrated the presence of multilayers.²⁵ Although multilayering appears to be an intrinsic property of ILs at all charged interfaces, the geometric constraints of a solid surface are likely to enhance it.

In the present study, we measure XR from the soft IL/W interfaces under potential control by a four-electrode system and characterize the potential dependence of the EDL structure of ILs.

METHODS

Materials. We used a hydrophobic ionic liquid, trihexyltetradecylphosphonium bis(nonafluorobutanesulfonyl)amide⁵⁶ ([THTDP⁺][C₄C₄N[−]]), which forms a well-defined interface with an aqueous solution of LiCl. Figure 1a illustrates its structure. [THTDP⁺][C₄C₄N[−]] was prepared from [THTDP⁺]Cl[−] (Aldrich) and Li⁺[C₄C₄N[−]] (Mitsubishi Materials Electronic Chemicals) and purified using the same method for a hydrophobic IL trioctylmethylammonium C₄C₄N[−] described elsewhere.⁵⁹ The density of [THTDP⁺][C₄C₄N[−]] is 1.213 g cm^{−3} (=0.382 e Å^{−3}) at 25 °C.⁵⁶

Electrochemical System. A four-electrode electrochemical system, shown in Figure 1b, is used to polarize the IL/W interface.⁴⁸ Although this liquid/liquid system forms EDLs on both sides of the interface, our XR measurements are relatively insensitive to the presence of the LiCl EDL on the aqueous side for the following reasons. The Li⁺ and Cl[−] electrolytes in

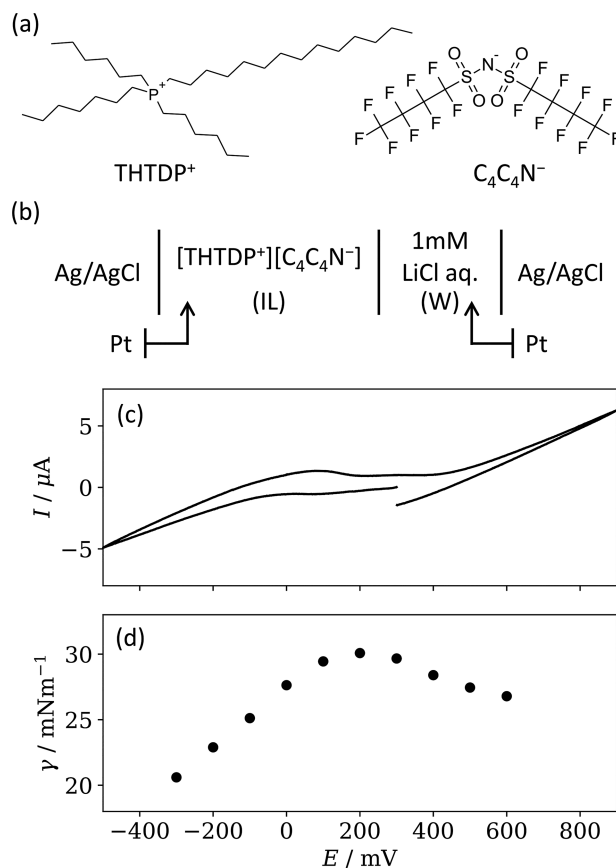


Figure 1. (a) Structure of [THTDP⁺][C₄C₄N[−]]. (b) Electrochemical system at the interface between [THTDP⁺][C₄C₄N[−]] and 1 mM LiCl aqueous solution. (c) Cyclic voltammogram at 10 mV s^{−1}. (d) Electrocapillary curve whose apex at roughly 200 mV corresponds to the potential of zero charge.

water do not specifically adsorb to the IL/W interface,^{60,61} and the low concentration of LiCl in the water (1 mM) produces a low X-ray contrast of the LiCl EDL in the water.

A cyclic voltammogram of the IL/W interface is shown in Figure 1c, which was taken using the setup for electrocapillary measurement described below with a positive-feedback *IR* compensation. The potential of Ag/AgCl in the aqueous phase with respect to that in the IL phase is denoted as *E*. The potential window of the IL/W interface was estimated to span the range from −300 to 600 mV. The negative limit of the potential window corresponds to ion transfer of Cl[−] from W to IL, and the positive limit corresponds to transfer of C₄C₄N[−] from IL to W, as determined by the standard ion-transfer potentials of THTDP⁺, C₄C₄N[−], Li⁺, and Cl[−] in the oil–water two-phase system.^{62,63}

Electrocapillary Measurement. The interfacial tension at the IL/W interface as a function of potential (known as the electrocapillary curve) was measured. As shown in Figure 1b, Ag/AgCl wires were used as the reference electrode in the water phase and the quasi-reference electrode in the IL. Platinum wires were used as the counter electrodes in the IL and water phases. The potential was controlled using a four-electrode potentiostat (Hokuto Denko, HA1010mM1A). A pendant drop of IL in the water phase was formed at the tip of a glass tube (inner diameter: 3 mm) that was filled with the IL. After applying the potential, the drop was held for 14 min to complete the EDL charging relaxation. Then, the picture of the

droplet was taken every 5 s for 1 min and the outlines of the droplet were extracted. The temperature was controlled at 25 °C. The details of the equipment and data analysis to determine the interfacial tension from the outline were described elsewhere.^{60,64}

From the electrocapillary plot (Figure 1d), PZC was estimated by the apex of the electrocapillary parabola to be roughly 200 mV ($=E_{\text{PZC}}$). The IL side of the interface is negatively charged when $E > E_{\text{PZC}}$ and positively charged when $E < E_{\text{PZC}}$.^{60,61}

X-ray Reflectivity Measurement and Data Analysis.

XR was performed using the liquid surface reflectometer at ChemMatCARS Sector 15 at Advanced Photon Source (APS, at Argonne National Laboratory) and the same electrochemical system shown in Figure 1b. A cell made of polyetheretherketone (PEEK) was designed to control the potential at the liquid–liquid interface (Figure 2). The

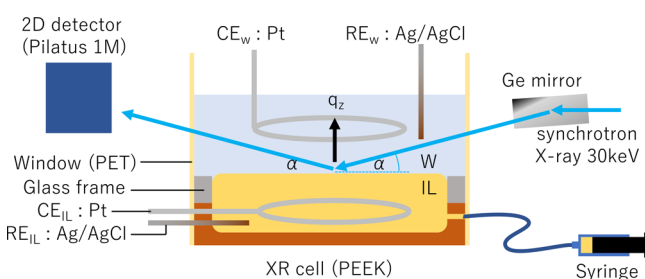


Figure 2. Schematic illustration of the XR measurement and electrochemical cell.

electrodes on the lower IL phase penetrate the cell side wall through two sealed lateral holes. To keep the IL/W interface flat, the volume of the lower IL phase was finely adjusted using a syringe connected through another sealed lateral hole. A quartz glass frame was placed inside the cell to make a flat interface. The glass frame was pretreated by being immersed in a saturated potassium hydroxide ethanol solution to make the surface hydrophilic and then coated with a 1% dimethyldichlorosilane–hexane solution only on the inner side surface to make it hydrophobic. This surface modification made the upper surface of the frame W-wet and the inner side surface IL-wet.⁵⁵ During the measurement, the potential was controlled using a four-electrode potentiostat (Solartron SI 1287).

X-ray reflectivity $R(q_z)$ was measured as a function of the wave vector transfer $q_z = 4\pi/\lambda \sin \alpha$, where λ is the wavelength of X-rays and α is the incident angle. We used 30 keV X-ray to reduce the absorption of the incident and reflected X-rays passing through the upper water phase. The measured $R(q_z)$ were normalized by the Fresnel reflectivity $R_F(q_z)$, which is a reflectivity calculated assuming an ideally flat and structureless interface between $[\text{THTDP}^+][\text{C}_4\text{C}_4\text{N}^-]$ and water. The reflected X-rays were detected by a two-dimensional detector (Pilatus 1M). Details of the XR measurements were described elsewhere.⁶⁵ The measurement was performed at room temperature.

The electron density distribution $\rho(z)$ as a function of the direction normal to the interface that reproduces experimental R/R_F was determined based on the model fitting. Note that only the EDL on the IL side of the interface is important because 1 mM LiCl aq, which surely forms EDL on the water side of the interface, is too dilute to affect the shape of the $\rho(z)$. As described above, neither Li^+ nor Cl^- forms an

adsorbed layer at the IL/W interface.^{60,61} Also, the ionic diffuse layer on the water side has a low X-ray contrast because of such a low concentration of these relatively low-electron-density elements. Therefore, by neglecting the EDL on the water side and considering ionic layers on the IL side, the $\rho(z)$ profile can be described as follows using a slab model

$$\rho(z) = \rho_0 + \sum_{i=1}^{N+1} (\rho_i - \rho_{i-1}) \frac{1 + \text{erf}\left(\frac{z - z_i}{\sqrt{2}\sigma}\right)}{2} \quad (1)$$

where $\text{erf}(x) = 2/\sqrt{\pi} \int_0^x \exp(-t^2) dt$ is the error function.

The variable N is the number of interface layers, and two models with $N = 1$ and 2 were used and referred to as “one-slab model” and “two-slab model”, respectively, in the present study. In these models, $\rho(z)$ is expressed with the thickness $d_i = z_{i+1} - z_i$ and the electron density ρ_i of the interface layer i , and the interface roughness σ . The electron density of the aqueous phase ($i = 0$) and the IL phase ($i = N + 1$) was estimated to be $\rho_0 = \rho_W = 0.334$ and $\rho_{N+1} = \rho_{\text{IL}} = 0.382 \text{ e } \text{\AA}^{-3}$, respectively, from the molecular weights, densities,⁵⁶ and electron numbers. R/R_F was numerically calculated from $\rho(z)$ by using the Parratt method.⁶⁶ The model parameters were fitted to minimize the residual sum of squares between the calculated and experimental R/R_F using a nonlinear least-squares method. It is meaningful to compare the roughness σ of the slab model with the interface roughness σ_{CWT} evaluated by the capillary wave theory. σ_{CWT} is expressed in the following equation⁶⁷

$$\sigma_{\text{CWT}}^2 = \frac{k_B T}{2\pi\gamma} \ln \left(\frac{q_{\text{max}}}{q_{\text{res}}} \right) \quad (2)$$

where k_B is the Boltzmann constant, T is the absolute temperature, and γ is the interfacial tension that is measured using the electrocapillary measurements described above. The value of q_{max} is the maximum wavenumber determined by the size of the molecule and is estimated as $q_{\text{max}} = 2\pi/D_{\text{mol}}$. $D_{\text{mol}} = 15 \text{ \AA}$ was approximated from the thickness of the ionic layer obtained from our previous XR study at the surface of $[\text{THTDP}^+][\text{C}_4\text{C}_4\text{N}^-]$.⁵⁶ The value of q_{res} is the smallest measurable wavenumber determined by the XR instrument and is estimated as $q_{\text{res}} \sim (h/L)q_d$, where $h = 1.892 \text{ mm}$ is the height of the detector slit and $L = 2.655 \text{ m}$ is the distance from the center of the interface to the detector.⁶⁵

Model Selection for the Electron Density Profile. At first, we used the one-slab model for the electron density profile. The one-slab model has the following three parameters: the electron density ρ and thickness d of the slab and the interfacial roughness σ . The results of the fitting to the $R/R_F(q_z)$ and the modeled electron density profiles are shown in Figure S2, and the best-fit parameters are listed in Table S1 and shown in Figure S1a,c,e. The one-slab model was able to reproduce the $R/R_F(q_z)$ plots at all of the potentials except for 450 and 550 mV. However, the best-fit parameters were physically unreasonable. At the potentials around PZC, the thickness of the slabs (25–35 Å) was significantly larger than the size of the ions (Figure S1c). At the potentials below PZC, the roughness σ was larger than the capillary wave roughness σ_{CWT} (Figure S1e,f), which is the experimental upper limit in XR. These unreasonable parameters indicate that the one-slab model does not suit this system, although the model reproduces the shape of the electron density profile.

Then, we examined other models. The distorted crystal model,^{68,69} which was used for the previous XR studies for the IL/solid²⁰ and IL/air⁵⁵ interfaces, also reproduced the $R/R_F(q_z)$ plots. However, we decided not to use this model because it requires more fitting parameters and the $R/R_F(q_z)$ plots do not always show convex behavior characteristic to this model. Next, we extended the slab model from the one- to two-slab model. The extension increases the number of fitting parameters from three to five. As shown in Figure S2, three parameters are enough to reproduce the $R/R_F(q_z)$ plot. In order to prevent overfitting, we fixed σ to σ_{CWT} and used the same thickness for the first and second layers. Therefore, the fitting parameters for the two-slab model are the electron densities of the first slab (ρ_1) and the second slab (ρ_2) and the thickness of slabs ($d = d_1 = d_2$). The best-fit parameters of the two-slab model are listed in Table S2 and shown in Figure S1b,d. ρ_1 and ρ_2 show a clear and systematic tendency to be higher and lower as the potential becomes positive, crossing at $E = E_{\text{PZC}}$ (Figure S1b). d ($= d_1 = d_2$) was in the order of ion diameter (Figure S1d). The potential dependence (or independence) of the fitted parameters is physicochemically justifiable. The χ^2 values for the two-slab model are similar to those for the one-slab model at most potentials with significant improvement at the two most positive potentials (Tables S1 and S2). Therefore, we concluded that the two-slab model used here is appropriate to extract the information on the EDL structure from XR data, which is lost when using the one-slab model and also other models with more fitting parameters.

RESULTS AND DISCUSSION

X-ray Reflectivity. Measurements of normalized reflectivity R/R_F as a function of the wave vector transfer q_z for a range of potentials are shown in Figure 3. They provide a structural characterization of the interfacial EDL over the range of potentials $E - E_{\text{PZC}}$ varying from -450 to 350 mV. Peaks develop as the potential $E - E_{\text{PZC}}$ increases from 150 to 350 mV, for which the IL side of the interface is negatively charged.

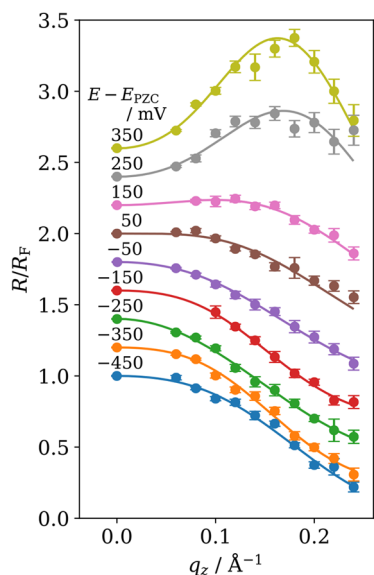


Figure 3. X-ray reflectivity R normalized by the Fresnel reflectivity R_F as a function of the surface-normal wave vector transfer q_z for several different potentials which are offset vertically for clarity. The solid lines are best-fit curves from the two-slab model described in the text.

Since the electron density of IL ($\rho_{\text{IL}} = 0.382 \text{ e} \text{ \AA}^{-3}$) is larger than that of W ($\rho_{\text{W}} = 0.334 \text{ e} \text{ \AA}^{-3}$), the peaks indicate the formation of an interfacial layer whose ρ is either higher than ρ_{IL} or lower than ρ_{W} . Considering that the fluorinated anion $\text{C}_4\text{F}_9\text{N}^-$ has a higher ρ than THTDP^+ ²⁵ and taking into account the polarity of the applied potential, the peaks reveal the formation of a high- ρ layer due to an enrichment of IL anion at the interface. On the other hand, the nearly monotonic R/R_F at potentials near and below the PZC suggests a nearly monotonic variation of the electron density profile, as verified by modeling the profile to fit the reflectivity data.

We examined several models, as described above, and found that a two-slab model reproduced the R/R_F for all potentials. The two-slab model consists of two uniform bulk phases (W and IL) that sandwich two slabs; each slab is a homogeneous layer of uniform electron density, as illustrated by the dashed lines in Figure 4. The internal interfaces in this model (W/first-

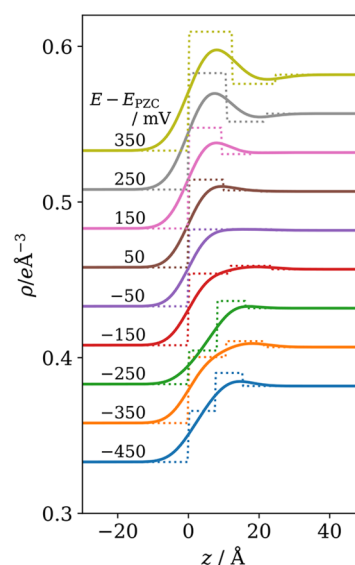


Figure 4. Electron density profiles $\rho(z)$ at the IL/W interface derived by fits to the data in Figure 3, where z is the displacement along the surface normal. Profiles at different potentials are offset for clarity. Bulk water is located at $z \leq 0$ and bulk IL at $z \geq 0$. The dotted lines represent the profiles of the two-slab model without capillary wave roughness.

slab, first-slab/second-slab, second-slab/IL) are roughened by capillary waves using a single value of roughness determined from the measured interfacial tension (Figure 1d) and capillary wave theory.^{67,70} As will be shown, the first and second slabs describe the layering of IL cations and anions within the EDL. Although the IL cations and anions are of slightly different sizes, they are nearly the same within the resolution of these measurements. Therefore, the thicknesses of the first and second slabs were set equal in order to reduce the number of free parameters in the model. Fitted R/R_F curves and profiles of ρ are shown in Figures 3 and 4, respectively, and the best-fit parameters, d , ρ_1 , and ρ_2 , are listed in Table S2, where d is the thickness and ρ_1 and ρ_2 are the electron densities of the first and second slabs.

Figure 4 illustrates the bilayer structure of the EDL. The fitted values of slab thickness d are comparable to the layer thickness measured previously of ionic multilayers at the

liquid/vapor interface of $[\text{THTDP}^+][\text{C}_4\text{C}_4\text{N}^-]$.⁵⁶ At $E - E_{\text{PZC}} \geq 50$ mV, the first slab, which represents the ionic layer adjacent to water ($0 \text{ \AA} \leq z \leq \sim 10 \text{ \AA}$), has a higher electron density ρ_1 than the IL bulk (see Table S2 for numerical values). Since $\text{C}_4\text{C}_4\text{N}^-$ has a higher ρ than THTDP^+ , the first layer is enriched with anions $\text{C}_4\text{C}_4\text{N}^-$, as expected for $E > E_{\text{PZC}}$. The electron density ρ_1 increases with increasing potential, revealing the accumulation of $\text{C}_4\text{C}_4\text{N}^-$ at the interface.

Over the same potential range (50–350 mV), ρ_2 is reduced to values below the IL bulk as the potential increases. This reveals cation accumulation in the second layer that accompanies anion accumulation in the first layer. These observations reveal the gradual development of alternately charged ionic multilayers in ILs, as suggested previously by MD^{13,14} and theoretical^{18,19} studies. The present result confirms the generality of alternately charged ionic multilayers at IL interfaces, not just at solid electrodes,^{20,22,24} where the hard surface provides a geometric constraint, but also at the softer electrochemical liquid/liquid interface, which fluctuates with capillary waves.

Figure 4 reveals that the thickness of the layers increases with the potential at positive potentials, especially from 250 to 350 mV. This result of the fitting is a direct consequence of the decrease in the q_z value of the X-ray reflectivity peak with increasing potential that is observed in Figure 3 for these two potentials. This indicates that electrostriction effects are not substantial in these samples, if present at all, because an electrostrictive thinning of the layers would yield peaks in XR that shift to larger values of q_z with increasing potential.

Figure 4 shows that the situation is reversed for negative potentials when $E - E_{\text{PZC}} \leq -150$ mV: ρ_1 and ρ_2 are respectively lower and higher than ρ_{IL} (Table S2), indicating the enrichment of cations and anions, respectively, within the first and second layers. As presented above, not only the two-slab model but also the one-slab model is able to reproduce the similar shape of $\rho(z)$. However, the one-slab model requires σ unreasonably larger than σ_{CWT} (Table S2) obtained from the electrocapillary measurement and therefore is not suitable as a model to discuss the interfacial structures. The variation of best-fit parameters from the two-slab model (Table S2) over this range of negative potentials was less systematic than that for positive potentials, primarily because of the nearly monotonic variation of the XR data. Since the electron density of the first layer is higher than water but lower than IL at these negative values of $E - E_{\text{PZC}}$, the electron density profiles are also nearly monotonic with only a weak peak appearing in the second layer. Nevertheless, fitting results at these negative potentials indicate that $\rho_1 < \rho_{\text{IL}} < \rho_2$, which strongly suggests the inversion of the cation- and anion-rich bilayers triggered by changing the sign of the potential.

The presence of alternately charged ionic multilayers was observed previously by experiment,^{20,22,24} MD simulation,^{13,14} and theory^{18,19} at IL/solid interfaces. Alternately charged layers imply that the surface charge density of the first ionic layer exceeds that of the electrode, resulting in overscreening which allows a second layer of opposite charge to form.^{18,19} Although simulation^{13,14} and theory^{18,19} predict that further charging of the electrode produces crowding of ions in the EDL, we did not observe crowding in this experiment. Since the XR technique would be very sensitive to the predicted structure of a crowded EDL,^{13,14} it is possible that higher potentials are required to achieve a crowded state.

Excess of Electrons. Further quantitative evaluation of the potential dependence of cation and anion distributions in the EDL is illustrated in Figure 5 by the electron excess within

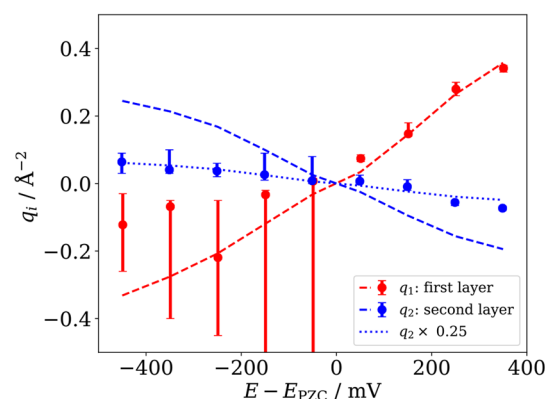


Figure 5. Electron excess in the first and second ionic layers, q_1 (red) and q_2 (blue), respectively, evaluated from the two-slab model fits to the X-ray reflectivity data. The dashed lines are q_1 (red) and q_2 (blue) from the BSK-GCS model, and the dotted line is $0.25q_2$ (see text and the Supporting Information for details).

each ionic layer, $q_i = (\rho_i - \rho_{\text{IL}})d$, where ρ_{IL} is the bulk electron density, ρ_i is the layer electron density, and d is the layer thickness. The electron excess measures the excess surface density of electrons within a layer over that within a similar volume of bulk IL. It should be noted that q_i is not the surface excess of ions frequently used in the electrochemical EDL analysis but still reflects the enrichment and the depletion of cations and anions at the ionic layers. Error bars in Figure 5 were determined by χ^2 mapping (Figures S3–S6). With increasing potential, q_1 increased and q_2 decreased, revealing the development of alternately charged multilayer structure. Error bars are large below the PZC because the R/R_F plots in this region are relatively featureless, as discussed previously. Nevertheless, it appears that $|q_1| > |q_2|$ at all potentials, illustrating the damped oscillatory behavior of the layer-by-layer ion enrichment expected for overscreening.

To compare with the experimental electron excess q_i , we used the alternately charged multilayer model proposed by Bazant, Storey, and Kornyshev,^{18,19} hereafter abbreviated as the BSK model, for the EDL in the IL phase, which was combined with the Gouy–Chapman–Stern (GCS) model with a 3 Å thick Stern layer for the EDL in the water phase. We refer the reader to the Supporting Information for a detailed description of this BSK–GCS model. This model improves upon a model applied in a previous XR study of the IL/W interface which did not take into account the overscreening effect.²⁵ We found that an accurate accounting of the electron excess shown in Figure 5 required us to account for the difference in the cation and anion size that was not taken into account in the BSK model. The effect of the size asymmetry is represented by ξ , the volume ratio of anion to cation. Although the BSK model^{18,19} incorporates the Kornyshev model³ to derive charge density distributions, we used the Oldham model,⁷¹ which treats the special case of the Kornyshev model in which compression of ions at the interface is not taken into account. The ionic compression led to a discrepancy in the ρ profile from the EDL model calculation and the XR result.

The electrostatic correlation length l_c in the BSK model was estimated to be 10 Å from the Debye length in this IL, $\lambda_D \sim$

1.0 Å.^{18,19} With the ion size asymmetry ξ set to be 0.75, the q_1 curve in Figure 5 matches the experimental values. This value of ξ is comparable to a size asymmetry of 0.58 that was estimated by DFT calculations of ionic volume at the level of B3LYP/6-311G**.⁵⁶ Since such calculations of a single ion in a vacuum neglect the interionic volume between cation and anion,⁷² ξ is probably underestimated in this DFT calculation.

The q_2 curve predicted by the same parameters (blue dashed line, Figure 5) did not agree with experimental results, suggesting that the BSK model overestimates ion enrichment in the second layer or, equivalently, underestimates the damping of the oscillatory layer-by-layer ion enrichment. Ion enrichment in the second layer is overestimated by a factor of 4, as demonstrated by multiplying q_2 by 0.25 (Figure 5), whose values then agree with experiment. Interestingly, we found good agreement for both q_1 and q_2 when the correlation length l_c is taken to be smaller than the ion size. However, this may be an unphysical artifact of the model, since it is unlikely that l_c would be smaller than the ion size in the case of a globular ion like THTDP⁺, though packing of the more rigid C₄C₄N[−] might lead to such an effect.

CONCLUSIONS

In summary, we investigated the potential dependence of the EDL structure at the electrified liquid/liquid interface between the ionic liquid [THTDP⁺][C₄C₄N[−]] and a LiCl aqueous solution by X-ray reflectivity. These data address the challenge posed by theory^{18,19} and MD simulation^{13,14} of measuring the gradual development of alternately charged interfacial layers as the electrical potential is varied. A C₄C₄N[−]-rich first layer and a THTDP⁺-rich second layer were observed at positive potentials. Although experimental uncertainties were larger at negative potentials, the formation of a THTDP⁺-rich first layer and a C₄C₄N[−]-rich second layer was consistent with our data at positive potentials. The present experimental results revealed the gradual development and change in polarity of alternately charged ionic multilayers upon applying a range of negative to positive potentials. Comparison of the excess electron surface density within each layer of the EDL with a model based upon current theory yielded good results for the density within the first layer but poor agreement with the density in the second layer, suggesting that further development of the EDL theory for ILs is required.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcb.0c03711>.

XR model fitting and EDL model calculations (PDF)

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Due to production error, eq 1 error function was misrepresented in the version published on July 14, 2020 and was corrected on July 23, 2020.