

## Comprehensive S/TEM Study of Interfaces in CVD Grown Vertical and In-plane Heterostructures of Two-Dimensional MoS<sub>2</sub> and ReS<sub>2</sub>

Saiphaneendra Bachu<sup>1</sup>, Lauren Stanton<sup>2</sup>, Chenhao Qian<sup>2</sup>, Danielle Reifsnyder Hickey<sup>1</sup> and Nasim Alem<sup>2</sup>

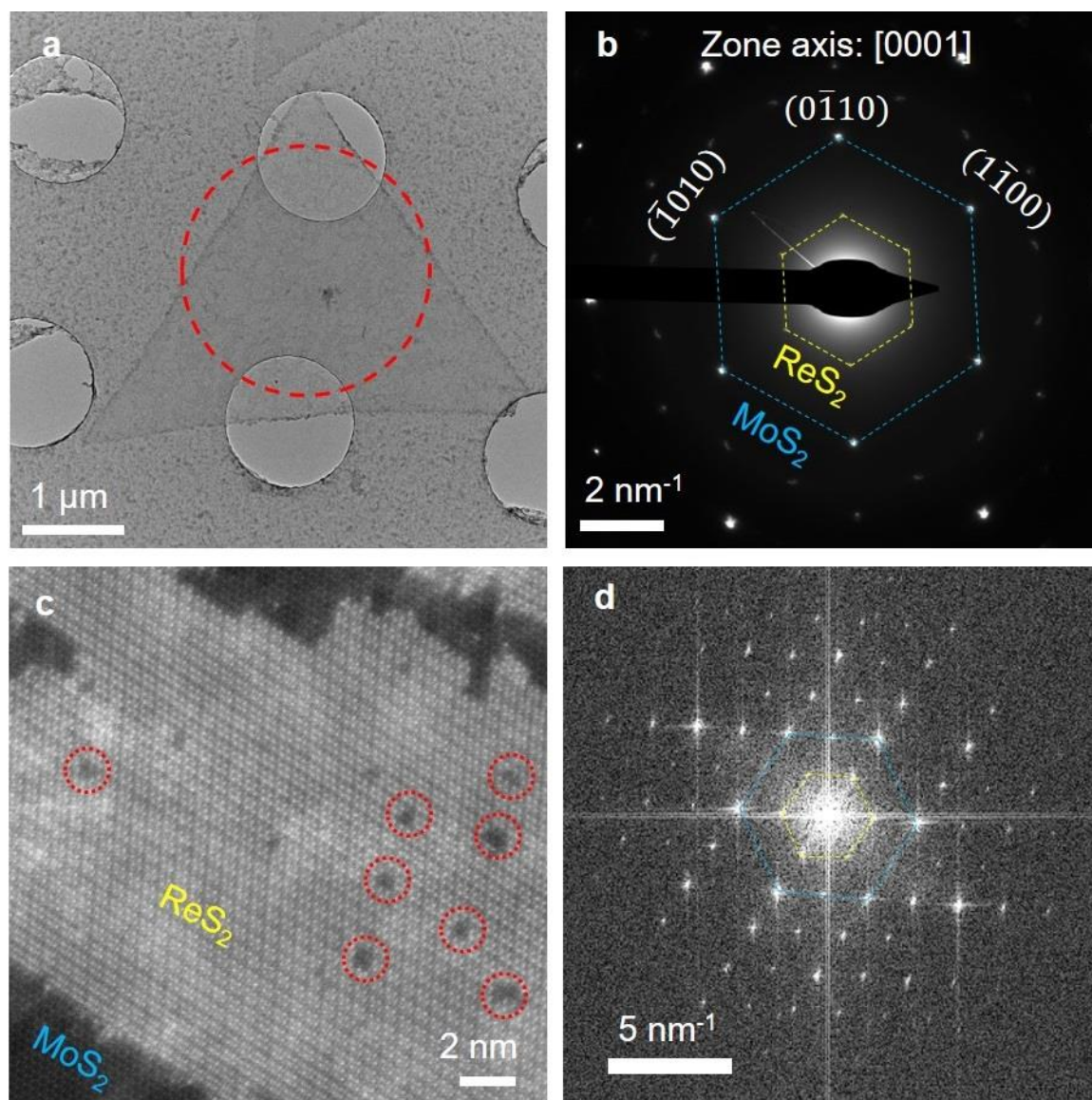
<sup>1</sup>Pennsylvania State University, State College, Pennsylvania, United States, <sup>2</sup>Pennsylvania State University, University Park, Pennsylvania, United States

Two-dimensional (2D) transition metal dichalcogenides (TMDs) are technologically consequential materials owing to their attractive properties such as indirect to direct band gap transition upon thinning to a monolayer [1]. Integration of multiple 2D TMDs into heterostructures in various geometries (vertical and in-plane) has accelerated the development of these materials into targeted applications in optoelectronics [2]. However, the performance of these heterostructures is highly dependent on the interfaces formed between constituent TMDs [3]. For example, defects and strain at the interface, originating from the lattice mismatch between the two 2D materials, can significantly affect the interface structure and consequently the heterostructure performance [4]. At the same time, this provides an opportunity to control the interface characteristics by careful selection of TMDs and synthesis methods to tune the properties of resulting heterostructures.

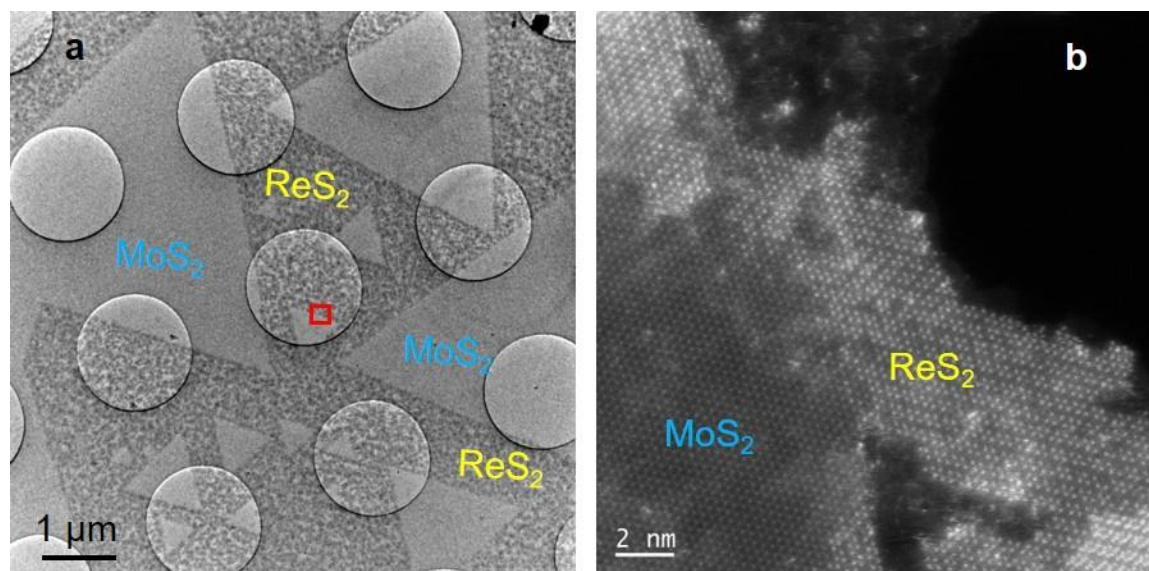
Most of the studies on heterostructures so far focused on those formed by isotropic TMDs such as MoS<sub>2</sub>, WS<sub>2</sub>, WSe<sub>2</sub> and MoSe<sub>2</sub>. Whereas, reports on heterostructures formed between an isotropic and an anisotropic TMDs are sparse. In this study, ReS<sub>2</sub> was chosen to form an interface with MoS<sub>2</sub> because those two materials have different crystal structures. ReS<sub>2</sub> has an anisotropic structure, as opposed to isotropic MoS<sub>2</sub> [5]. Such anisotropy in the system can introduce modulations in the atomic and interfacial structure that is derived from strain and lattice mismatch. Moreover, MoS<sub>2</sub>-ReS<sub>2</sub> heterostructures are expected to exhibit type I band alignment, strong interlayer interaction and demonstrate excellent photoresponse properties [6]. Therefore, it is imperative to synthesize MoS<sub>2</sub>-ReS<sub>2</sub> heterostructures with well-defined interfaces and to understand their atomic structure.

Here, we used a two-step CVD process to synthesize vertical and in-plane MoS<sub>2</sub>-ReS<sub>2</sub> heterostructures wherein, MoS<sub>2</sub> is grown on c-plane sapphire during the first step and ReS<sub>2</sub> is subsequently grown on MoS<sub>2</sub>/sapphire in the second step. The as grown heterostructures are transferred to Cu quantifoil holey carbon grids for studying their microstructure. We employed scanning/transmission electron microscopy (S/TEM) techniques to investigate the interface atomic structure, defects, epitaxy and strain between the MoS<sub>2</sub> and ReS<sub>2</sub> layers. Figure 1a shows a low magnification TEM image of MoS<sub>2</sub>-ReS<sub>2</sub> vertical heterostructure. Selected area diffraction pattern (SADP) obtained from the area outlined by the dashed circle in Figure 1a is presented in Figure 1b. SADP highlights that ReS<sub>2</sub> aligns epitaxially with MoS<sub>2</sub> as evident by the matching orientation of diffraction spots. Figure 1c is an atomic resolution high-angle annular dark-field (HAADF)-STEM image of a vertical heterostructure showing monolayer ReS<sub>2</sub> grown on top of monolayer MoS<sub>2</sub>. FFT pattern obtained from the atomic resolution image, as shown in Figure 1d, further confirms the epitaxy between MoS<sub>2</sub> and ReS<sub>2</sub> at atomic length scale. Figure 2a is a low magnification TEM image illustrating how ReS<sub>2</sub> grows in the gap between MoS<sub>2</sub> triangles during the second step of the synthesis procedure leading to the formation of in-plane heterostructure. An atomic

resolution HAADF-STEM image taken from the edge of a MoS<sub>2</sub> triangle confirms the formation of in-plane heterostructure, as presented in Figure 2b. The presentation will include more in-depth analysis on various features observed in the heterostructures along with results from other TEM characterization techniques [7].



**Figure 1.** MoS<sub>2</sub>-ReS<sub>2</sub> vertical heterostructures: (a) Low magnification TEM image of MoS<sub>2</sub>/ReS<sub>2</sub> vertical heterostructure, (b) selected area DP obtained from the area outlined by the dashed circle in (a), (c) atomic resolution HAADF-STEM image of the vertical heterostructure showing the underlying monolayer MoS<sub>2</sub> with a monolayer ReS<sub>2</sub> crystal grown on top and (d) binned FFT pattern obtained from (c) highlighting the epitaxy between MoS<sub>2</sub> and ReS<sub>2</sub>.



**Figure 2.** MoS<sub>2</sub>-ReS<sub>2</sub> in-plane heterostructures: (a) Low-magnification TEM image showing MoS<sub>2</sub> triangles with ReS<sub>2</sub> grown between the triangles thus filling the space and (b) atomic-resolution HAADF-STEM image of in-plane heterostructure formed at the edge of a MoS<sub>2</sub> triangle, obtained from the region marked by the red box in (a).

#### References

- [1] Splendiani, A. et al., *Nano Lett.* **10**, 1271–1275 (2010).
- [2] Geim, A. K. & Grigorieva, I. V., *Nature* **499**, 419–425 (2013).
- [3] Pant, A. et al., *Nanoscale* **8**, 3870–3887 (2016).
- [4] Xie, S. et al., *Science* **359**, 1131–1136 (2018).
- [5] Lin, Yung-Chang, et al., *ACS Nano* **9.11**, 11249–11257 (2015).
- [6] Bellus, M. Z. et al., *Nanoscale Horiz.* **2**, 31–36 (2017).
- [7] This work was supported by the National Science Foundation (NSF), in part under the CAREER program (DMR-1654107), in part by the program EFRI 2-DARE: 2D Crystals by Activated Atomic Layer Deposition (EFRI-1433378), and in part by the Penn State 2D Crystal Consortium-Materials Innovation Platform (2DCC-MIP) under NSF cooperative agreement DMR-1539916.

