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Gd³⁺-activated narrowband ultraviolet-B persistent luminescence through persistent energy transfer

This research demonstrates the design of Gd³⁺-activated narrow-band ultraviolet-B persistent phosphors from Pr³⁺-, Pb²⁺- or Bi³⁺-activated ultraviolet-C or ultraviolet-B persistent phosphors through persistent energy transfer from Pr³⁺, Pb²⁺ or Bi³⁺ ions to Gd³⁺ ions. The Gd³⁺-activated narrow-band ultraviolet-B persistent phosphors may act as self-sustained glowing sources for dermatological therapy.



As featured in:



rsc.li/dalton Registered charity number: 207890

Dalton Transactions

PAPER

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Cite this: Dalton Trans., 2021, 50, 3499

Received 13th January 2021, Accepted 15th February 2021 DOI: 10.1039/d1dt00120e

Introduction

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Ultraviolet-B (UVB) spectral band refers to the spectrum of light in the range of 280-315 nm. UVB light in the natural solar spectrum has been used for centuries in the treatment of various skin disorders such as psoriasis and vitiligo.^{1,2} Action spectrum studies of UVB wavelengths for psoriasis carried out in the late 1970s revealed that it is the narrowband UVB (NB-UVB; 310-313 nm) emission that is most effective for phototherapy.³⁻⁵ These findings quickly led to the development and use of artificial NB-UVB light sources for dermatological therapy.⁶⁻¹⁰ Currently, phototherapy based on Philips TL-01 NB-UVB fluorescent tubes is the treatment of choice for various skin disorders which affect millions of people around the world.^{11,12}

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This work reports the realization of Gd³⁺ persistent luminescence in the narrowband ultraviolet-B (NB-UVB; 310-313 nm) through persistent energy transfer from a sensitizer of Pr^{3+} , Pb^{2+} or Bi^{3+} . We propose a general design concept to develop Gd³⁺-activated NB-UVB persistent phosphors from Pr³⁺-, Pb²⁺- or Bi³⁺-activated ultraviolet-C (200–280 nm) or ultraviolet-B (280–315 nm) persistent phosphors, leading to the discovery of ten Gd³⁺ NB-UVB persistent phosphors such as Sr₃Gd₂Si₆O₁₈:Pr³⁺, $Sr_3Gd_2Si_6O_{18}$: Pb²⁺ and $Y_2GdAl_2Ga_3O_{12}$: Bi³⁺ as well as five ultraviolet-B persistent phosphors such as Y₃Al₂Ga₃O₁₂:Pr³⁺, Sr₃Y₂Si₆O₁₈:Pb²⁺ and Y₃Al₂Ga₃O₁₂:Bi³⁺. The persistent energy transfer from the sensitizers to Gd³⁺ is very efficient and the Gd³⁺ NB-UVB afterglow can last for more than 10 hours. This study expands the persistent luminescence research to the NB-UVB as well as the broader ultraviolet-B spectral regions. The NB-UVB persistent phosphors may act as self-sustained glowing NB-UVB radiation sources for dermatological therapy.

> In developing phosphors for NB-UVB fluorescent tubes, Gd^{3+} is consistently used as the emitter,^{13–17} because the Gd^{3+} ${}^{6}P_{7/2} \rightarrow {}^{8}S_{7/2}$ emission transition gives efficient line emission at 311 nm, which is within the action spectrum of phototherapy. However, due to the forbidden nature of the transitions within the Gd³⁺ 4f⁷ configuration, Gd³⁺ has very poor optical absorption of 200-300 nm excitation light. To make the Gd³⁺activated phosphors "excitable" by the short-wavelength UV light, a sensitizer is needed. The role of the sensitizer is to absorb the excitation energy and to transfer it to Gd^{3+} . Accordingly, the emitting levels of the sensitizer should be resonant with the excited states (*i.e.*, ${}^{6}I_{I}$ or ${}^{6}P_{I}$ states) of Gd³⁺. The available sensitizers for Gd³⁺ photoluminescence include Pr³⁺ $(e.g., Y_2GdAl_5O_{12}:Pr^{3+}, GdBO_3:Pr^{3+}, LaPO_4:Pr^{3+}, Gd^{3+}), 1^{3-16}Pb^{2+}$ (e.g., GdBO₃:Pb²⁺),¹⁷ Bi³⁺ (e.g., LaB₃O₆:Bi³⁺,Gd³⁺; the phosphor used in the Philips TL-01 fluorescent tubes),⁶ and Ce^{3+} (e.g., $GdMgB_5O_{10}:Ce^{3+}$).¹⁸ Among these sensitizers, the ${}^{6}I_{I}$ or ${}^{6}P_{I}$ excited states of Gd³⁺ overlap with the 4f5d emitting level of Pr^{3+} , the ${}^{3}P_{1}$ emitting level of Pb^{2+} , the ${}^{3}P_{1}$ emitting level of Bi^{3+} , or the 5d emitting level of Ce^{3+} .

> The Gd³⁺-activated NB-UVB phosphors developed by far are photoluminescent, where continuous external excitation is needed. Here, we report the design and fabrication of a new type of Gd³⁺-activated NB-UVB phosphors, *i.e.*, Gd³⁺-activated NB-UVB persistent phosphors, which can emit Gd³⁺ 311 nm persistent luminescence (also called afterglow) without the need of constant external excitation. For a persistent phosphor to work, the host should contain appropriate energy traps that can capture and store the excitation energy (in the form of



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[†]Electronic supplementary information (ESI) available: XRD patterns of NB-UVB persistent phosphors Sr₃Gd₂Si₆O₁₈:Pr³⁺, Sr₃Gd₂Si₆O₁₈:Pb²⁺, Y₂GdAl₂Ga₃O₁₂:Pr³⁺ and Y2GdAl2Ga3O12:Bi3+, persistent luminescence properties of NB-UVB persistent phosphors Ca_{1.8}Gd_{0.2}Al₂SiO₇:Pr³⁺, Lu_{1.8}Gd_{0.2}SiO₅:Pr³⁺, Ca_{2.8}Gd_{0.2}Al₂Si₃O₁₂: $Pr^{3+}, \quad LiY_{0.8}Gd_{0.2}SiO_4:Pr^{3+}, \quad Sr_3Y_{2-x}Gd_xSi_6O_{18}:Pb^{2+}, \quad Lu_2GdAl_2Ga_3O_{12}:Pr^{3+} \quad and \quad Sr_3Y_{2-x}Gd_xSi_6O_{18}:Pb^{2+}, \quad S$ $Lu_2GdAl_2Ga_3O_{12}{:}Bi^{3+}\text{, persistent luminescence properties of UVB persistent}$ phosphors Lu₃Al₂Ga₃O₁₂:Pr³⁺ and Lu₃Al₂Ga₃O₁₂:Bi³⁺, and conceptual self-sustained NB-UVB Band-Aid phototherapy. See DOI: 10.1039/d1dt00120e ‡These authors contributed equally to this work.

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electrons through the excitation of the activator) during the excitation and, after the excitation is ceased, gradually release the trapped electrons to the activator to produce persistent luminescence.^{19,20} For Gd³⁺, however, since it has a stabilized 4f⁷ configuration, it has no tendency to be oxidized or reduced in a phosphor. Consequently, there is no electron transfer between Gd³⁺ and the trapping states, and therefore Gd³⁺ cannot directly play the role of persistent luminescence emitting center in a persistent phosphor.²¹ Since Pr³⁺, Pb²⁺, Bi³⁺ and Ce³⁺ are efficient steady-state sensitizers for Gd³⁺ photoluminescence in some hosts,^{6,13–18} we propose to expand their use as persistent sensitizers to persistently transfer energy to Gd³⁺ to produce Gd³⁺ NB-UVB persistent luminescence (note that persistent energy transfer strategy was previously used to design and fabricate visible- and infrared-emitting persistent phosphors).²²⁻²⁴ Taking Pr³⁺ as the persistent sensitizer, this persistent energy transfer process can be illustrated by the schematic Pr³⁺-Gd³⁺ energy transfer diagram in Fig. 1. Besides the sensitizer function, Pr³⁺ should also function as a persistent luminescence activator. However, instead of giving out its own persistent emission in the ultraviolet-C (UVC; 200–280 nm) or the UVB via the Pr^{3+} 4f5d \rightarrow $^{3}H_{4,5}$ emission transitions, the energy released to the Pr³⁺ 4f5d emitting state from the trapping states is continuously and efficiently transferred to the Gd^{3+} ⁶I₁ levels (~35700–37000 cm⁻¹, corresponding to ~280-270 nm) or the lower-energy ${}^{6}P_{I}$ levels (~31 700-33 200 cm⁻¹, corresponding to ~315-301 nm), sub-

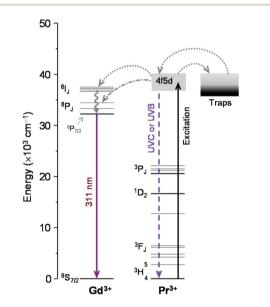


Fig. 1 Schematic representation of persistent energy transfer from Pr^{3+} to Gd^{3+} in a NB-UVB persistent phosphor. Under high-energy UV light (*e.g.*, 254 nm light) excitation, Pr^{3+} ions absorb the excitation energy which is then stored (in the form of electrons) in the energy traps in the material. After the excitation is ceased, the captured energy is thermally released to the Pr^{3+} 4f5d emitting level, which is then persistently transferred to the Gd^{3+} $^{6}I_{J}$ levels or $^{6}P_{J}$ levels, subsequently producing persistent emission at 311 nm *via* the Gd^{3+} $^{6}P_{J/2} \rightarrow {}^{8}S_{7/2}$ emission transition. Straight-line arrows and curved-line arrows represent optical transitions and electron transfer processes, respectively.

sequently producing Gd^{3+} persistent luminescence at 311 nm via the $Gd^{3+} {}^{6}P_{7/2} \rightarrow {}^{8}S_{7/2}$ emission transition. Therefore, we propose that a Gd^{3+} -activated, Pr^{3+} -sensitized NB-UVB persistent phosphor can be designed upon a Pr^{3+} -activated UVC persistent phosphor or a Pr^{3+} -activated UVB persistent phosphor. This design principle is also valid for other sensitizers of Gd^{3+} .

Recently, we developed five types of Pr³⁺-activated, silicatebased UVC persistent phosphors that emit long-time persistent luminescence peaking at 265-270 nm.²⁵ These Pr³⁺-activated UVC persistent phosphors provide us an ideal platform to test the Pr³⁺ to Gd³⁺ persistent energy transfer and our concept of designing Gd³⁺-activated NB-UVB persistent phosphors. Here we use one type of UVC persistent phosphor, the cyclosilicate structured Sr₃Y₂Si₆O₁₈:Pr³⁺, as the benchmark material to demonstrate the design of a Gd³⁺-activated, Pr³⁺-sensitized NB-UVB persistent phosphor, *i.e.*, Sr₃Gd₂Si₆O₁₈:Pr³⁺. To further demonstrate the universality of the design concept, we also specially developed five first-ever Pr3+-, Pb2+- and Bi3+-activated UVB persistent phosphors, represented by $Y_3Al_2Ga_3O_{12}$:Pr³⁺, Sr₃Y₂Si₆O₁₈:Pb²⁺ and Y₃Al₂Ga₃O₁₂:Bi³⁺, respectively, and used them to fabricate Gd³⁺ NB-UVB persistent phosphors $Y_2GdAl_2Ga_3O_{12}$: Pr^{3+} , $Sr_3Gd_2Si_6O_{18}$: Pb^{2+} and $Y_2GdAl_2Ga_3O_{12}$: Bi³⁺. In these NB-UVB persistent phosphors, the persistent energy transfer from the sensitizers to Gd³⁺ is very efficient and the Gd³⁺ NB-UVB afterglow can last for more than 10 h.

Experimental

Materials synthesis

All the phosphors discussed in this work were fabricated using a solid-state reaction method. Taking the synthesis of $Sr_3Gd_2Si_6O_{18}$: Pr^{3+} phosphor as an example, stoichiometric amounts of $SrCO_3$, Gd_2O_3 , SiO_2 and Pr_6O_{11} powders (Pr^{3+} content, 1 atom%) were ground to form a homogeneous fine powder. The mixed powder was prefired at 900 °C in air for 2 h. The prefired powder was ground again and was then pressed into 15 mm-diameter discs using a hydraulic press. The disc samples were then sintered at 1250 °C in air for 2 h to form solid ceramic discs.

Characterization of crystal structures

The X-ray diffraction patterns (XRD) of some phosphors were recorded using a PANalytical X'Pert PRO powder X-ray diffractometer with Cu K α 1 radiation (λ = 1.5406 Å).

Spectroscopic measurements

The steady-state photoluminescence properties of the phosphors were recorded using a McPherson spectrometer, which comprises of a McPherson Model 234/302 excitation monochromator and a McPherson Model 2035 emission monochromator. The persistent luminescence properties were measured using a Horiba FluoroLog-3 spectrofluorometer equipped with a R928P photomultiplier tube. An Ocean Optics QEPRO spectrometer was also used to record the persistent luminescence emission spectra. In decay-related measurements, a 254 nm mercury lamp was used to charge the samples.

Results and discussion

We firstly discuss the Sr₃Gd₂Si₆O₁₈:Pr³⁺ NB-UVB persistent phosphor developed from the Sr₃Y₂Si₆O₁₈:Pr³⁺ UVC persistent phosphor. The as-synthesized Sr₃Gd₂Si₆O₁₈:Pr³⁺ compound has the cyclosilicate structure, which shares the same structure as the benchmark compound Sr3Y2Si6O18 (ICDD 00-065- $(0204)^{26}$ (Fig. S1a, ESI[†]). In the Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor, Pr³⁺ ions occupy the Sr/Gd sites with eightfold coordination.²⁵ Fig. 2a shows the normalized photoluminescence excitation and emission spectra of Sr₃Gd₂Si₆O₁₈:Pr³⁺, Sr₃Y₂Si₆O₁₈:Pr³⁺ and Sr₃Gd₂Si₆O₁₈ phosphors. The spectra of the Gd³⁺-free Sr₃Y₂Si₆O₁₈:Pr³⁺ phosphor and the Pr³⁺-free Sr₃Gd₂Si₆O₁₈ phosphor are presented to emphasize the sensitizer role of Pr³⁺ in the Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor. Under excitation at 249 nm, the Sr₃Y₂Si₆O₁₈:Pr³⁺ phosphor exhibits a strong emission band peaking at ~266 nm and a weak shoulder band peaking at ~308 nm (the blue dash line curve), which can be attributed to the $Pr^{3+} 4f5d \rightarrow {}^{3}H_{4,5}$ and $4f5d \rightarrow {}^{4}H_{6}, {}^{3}F_{2}$ emission transitions, respectively. No apparent $Pr^{3+} 4f^2 \rightarrow 4f^2$ intraconfigurational emission transitions in the visible region are observed. The excitation spectrum monitored at 266 nm emission has a strong broad band peaking at ~249 nm (the magenta dot-dash line curve), which is originated from the $Pr^{3+3}H_4 \rightarrow 4f5d$ excitation transition. The spectral results show that the Sr₃Y₂Si₆O₁₈:Pr³⁺ phosphor is a good UVC photoluminescent material.

When the Y^{3+} ions in the $Sr_3Y_2Si_6O_{18}$: Pr^{3+} phosphor are substituted by Gd³⁺ ions, the resulting Sr₃Gd₂Si₆O₁₈:Pr³⁺ material becomes a pure NB-UVB phosphor. Under excitation at 249 nm, the material exhibits a strong, narrow emission band at 311 nm originating from the $\text{Gd}^{3+6}P_{7/2} \rightarrow {}^8S_{7/2}$ emission transition (the red solid line curve). No $Pr^{3+} 4f5d \rightarrow {}^{3}H_{4,5}$ and 4f5d $\rightarrow {}^{4}H_{6}, {}^{3}F_{2}$ emission transitions appear in the emission spectrum. The excitation spectrum monitored at the Gd³⁺ 311 nm emission consists of two excitation bands: one strong broad band peaking at 247 nm and one small narrow band peaking at 275 nm (the black solid line curve), which can be, respectively, attributed to the $Pr^{3+3}H_4 \rightarrow 4f5d$ and $Gd^{3+8}S_{7/2} \rightarrow$ ⁶I₁ excitation transitions. The photoluminescence results clearly show the occurrence of highly efficient energy transfer from Pr³⁺ to Gd³⁺ in the Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor. This efficient energy transfer is well reflected in the photoluminescence excitation and emission spectra of the two phosphors. For instance, the complete spectral overlap of the Gd³⁺ ${}^{8}S_{7/2} \rightarrow {}^{6}I_{I}$ excitation band of $Sr_{3}Gd_{2}Si_{6}O_{18}$:Pr³⁺ and the Pr³⁺ $4f5d \rightarrow {}^{3}H_{4,5}$ emission band of $Sr_{3}Y_{2}Si_{6}O_{18}$:Pr³⁺, the appearance of the $Pr^{3+3}H_4 \rightarrow 4f5d$ excitation band in the excitation spectrum of $Sr_3Gd_2Si_6O_{18}$: Pr^{3+} by monitoring the Gd^{3+} emission, the appearance of the $\text{Gd}^{3+}{}^6\text{P}_{7/2} \rightarrow {}^8\text{S}_{7/2}$ emission transition in the emission spectrum of Sr₃Gd₂Si₆O₁₈:Pr³⁺ by exciting the Pr³⁺ excitation band, as well as the absence of the Pr³⁺ $4f5d \rightarrow {}^{3}H_{4,5}$ UVC emission transitions when exciting the Pr^{3+} excitation band in Sr₃Gd₂Si₆O₁₈:Pr³⁺, all suggest the occur-

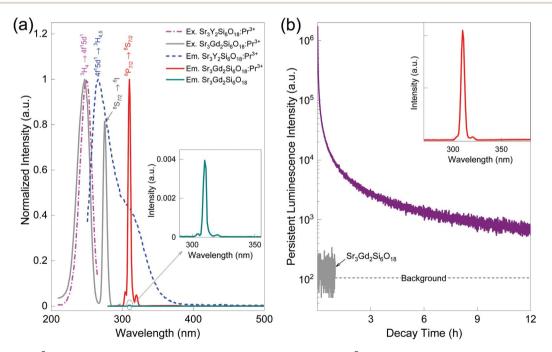


Fig. 2 $Sr_3Gd_2Si_6O_{18}:Pr^{3+}$ NB-UVB persistent phosphor developed from $Sr_3Y_2Si_6O_{18}:Pr^{3+}$ UVC persistent phosphor. (a) Normalized photoluminescence excitation and emission spectra of $Sr_3Gd_2Si_6O_{18}:Pr^{3+}$ NB-UVB persistent phosphor, $Sr_3Y_2Si_6O_{18}:Pr^{3+}$ UVC persistent phosphor and Pr^{3+} -free $Sr_3Gd_2Si_6O_{18}$ host at room temperature. The emission spectra are acquired under 249 nm light excitation and the excitation spectra are obtained by monitoring the 266 nm emission for the $Sr_3Y_2Si_6O_{18}:Pr^{3+}$ phosphor and the 311 nm emission for the $Sr_3Gd_2Si_6O_{18}:Pr^{3+}$ phosphor. The inset is the enlargement of the Gd^{3+} photoluminescence in $Sr_3Gd_2Si_6O_{18}$ host. (b) Persistent luminescence decay curves of $Sr_3Gd_2Si_6O_{18}:Pr^{3+}$ phosphor and $Sr_3Gd_2Si_6O_{18}$ host monitored at 311 nm after irradiation by a 254 nm lamp for 5 min. The inset shows the persistent luminescence emission spectrum of $Sr_3Gd_2Si_6O_{18}:Pr^{3+}$ phosphor recorded at 10 min decay.

Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor.

rence of efficient energy transfer from the Pr³⁺ 4f5d emitting state to the $Gd^{3+} {}^{6}I_{I}$ levels in the $Sr_{3}Gd_{2}Si_{6}O_{18}$: Pr^{3+} phosphor. the Gd³⁺ persistent luminescence in the Sr₃Gd₂Si₆O₁₈:Pr³⁺ Moreover, the sensitizer role of Pr³⁺ in the Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor.

phosphor is further underlined by the very weak Gd³⁺ photo-

luminescence in the Pr³⁺-free Sr₃Gd₂Si₆O₁₈ phosphor (inset of

Fig. 2a), where the Gd³⁺ photoluminescence emission intensity

is about three orders of magnitude lower than that of

in photoluminescence, efficient persistent energy transfer

from Pr³⁺ to Gd³⁺ also occurs after the excitation is ceased,

demonstrating that the Sr₃Gd₂Si₆O₁₈:Pr³⁺ phosphor has the

NB-UVB persistent luminescence capability. Fig. 2b shows the persistent luminescence decay curve of Sr₃Gd₂Si₆O₁₈:Pr³⁺

phosphor monitored at 311 nm after charged by a 254 nm

lamp. The data were recorded as a function of the persistent

luminescence intensity versus time and the recording lasted

for 12 h. The persistent luminescence intensity decreases

quickly in the first one hour and slowly afterward. After 12 h decay, the intensity is still over one order of magnitude higher

than the background signal. The inset in Fig. 2b shows the

persistent luminescence emission spectrum acquired at

10 min decay. Like the photoluminescence emission spectrum

in Fig. 2a, only the characteristic Gd³⁺ line emission appears

in the persistent luminescence emission spectrum. In stark contrast, no Gd³⁺ persistent luminescence signal is detected in

the Pr³⁺-free Sr₃Gd₂Si₆O₁₈ phosphor (Fig. 2b), which strongly

In addition to the steady-state Pr³⁺ to Gd³⁺ energy transfer

The above design method is general and can be used to fabricate more other Gd3+-activated, Pr3+-sensitized NB-UVB persistent phosphors based on existing Pr³⁺-activated UVC persistent phosphors. By incorporating certain amount of Gd³⁺ ions into four other types of Pr³⁺-activated, silicate-based UVC persistent phosphors in ref. 25, represented by the melilite structured Ca₂Al₂SiO₇:Pr³⁺, oxyorthosilicate structured Lu₂SiO₅:Pr³⁺, garnet structured Ca₃Al₂Si₃O₁₂:Pr³⁺ and orthosilicate structured LiYSiO₄:Pr³⁺, we, respectively, obtained NB-UVB persistent phosphors of Ca1.8Gd0.2Al2SiO7:Pr3+, Lu1.8Gd0.2SiO5:Pr3+, $Ca_{2,8}Gd_{0,2}Al_2Si_3O_{12}:Pr^{3+}$ and $LiY_{0,8}Gd_{0,2}SiO_4:Pr^{3+}$ (Fig. S2, ESI†).

For the Gd³⁺ NB-UVB persistent phosphors developed from Pr³⁺-activated UVC persistent phosphors, the persistent energy transfer is from the Pr^{3+} 4f5d emitting state to the $Gd^{3+} {}^{6}I_{I}$ energy levels. Equally important as this energy transfer route, according to Fig. 1, is persistent energy transfer from the Pr³⁺ 4f5d emitting state to the lower energy ${}^{6}P_{I}$ levels of Gd³⁺. This, however, requires the Pr³⁺-activated benchmark phosphor to emit in the UVB region with the Pr³⁺ 4f5d emitting state overlapping with the $Gd^{3+6}P_I$ levels. To test this idea, we first developed a Pr³⁺-activated UVB persistent phosphor Y₃Al₂Ga₃O₁₂: Pr³⁺, and then substituted one-third of its Y³⁺ ions with Gd³⁺ ions, forming a garnet-structured Y₂GdAl₂Ga₃O₁₂:Pr³⁺ phos-

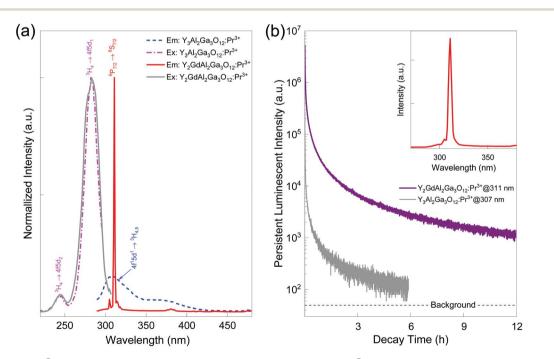


Fig. 3 Y_2 GdAl₂Ga₃O₁₂:Pr³⁺ NB-UVB persistent phosphor developed from Y_3 Al₂Ga₃O₁₂:Pr³⁺ UVB persistent phosphor. (a) Normalized photoluminescence excitation and emission spectra of Y_2 GdAl₂Ga₃O₁₂:Pr³⁺ and Y_3 Al₂Ga₃O₁₂:Pr³⁺ phosphors at room temperature. The emission spectra are acquired under 283 nm light excitation and the excitation spectra are obtained by monitoring the 307 nm emission for the Y₃Al₂Ga₃O₁₂:Pr³⁺ phosphor and the 311 nm emission for the Y_2 GdAl₂Ga₃O₁₂: Pr³⁺ phosphor. (b) Persistent luminescence decay curves of Y_3 Al₂Ga₃O₁₂: Pr³⁺ phosphor monitored at 307 nm and Y₂GdAl₂Ga₃O₁₂:Pr³⁺ phosphor monitored at 311 nm after irradiation by a 254 nm lamp for 5 min. The inset shows the persistent luminescence emission spectrum of Y2GdAl2Ga3O12:Pr3+ phosphor recorded at 10 min decay.

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phor, which has the same structure as the benchmark compound Y₃Al₂Ga₃O₁₂ (ICDD 04-007-4274)²⁷ (Fig. S1b, ESI[†]). Fig. 3a shows the normalized photoluminescence excitation and emission spectra of Y₃Al₂Ga₃O₁₂:Pr³⁺ and Y₂GdAl₂Ga₃O₁₂: Pr^{3+} phosphors. Under excitation at 283 nm, the Y₃Al₂Ga₃O₁₂: Pr³⁺ phosphor exhibits a broadband emission from Pr³⁺ peaking at ~307 nm originating from the Pr^{3+} 4f5d \rightarrow $^{3}H_{4,5}$ emission transitions (the blue dash line curve). This UVB emission band indicates that the energy of the Pr³⁺ 4f5d emitting state in the Y₃Al₂Ga₃O₁₂:Pr³⁺ phosphor matches that of the Gd^{3+ 6}P₁ levels; therefore, energy transfer from the Pr³⁺ 4f5d emitting state to the $Gd^{3+6}P_{I}$ levels is likely to occur in the Y2GdAl2Ga3O12:Pr3+ phosphor. Indeed, under excitation at 283 nm, the Pr^{3+} 4f5d \rightarrow $^{3}H_{4.5}$ emission transition in the $Y_2GdAl_2Ga_3O_{12}$:Pr³⁺ phosphor gives way to the Gd^{3+ 6}P_{7/2} \rightarrow ${}^{8}S_{7/2}$ emission transition, resulting in strong Gd³⁺ emission at

311 nm (the red solid line curve). The energy transfer from the Pr^{3+} 4f5d emitting state to the Gd³⁺ ${}^{6}P_{I}$ levels is further verified by the excitation spectra. The excitation spectrum of the Y₂GdAl₂Ga₃O₁₂:Pr³⁺ phosphor monitored at the Gd³⁺ 311 nm emission (the black solid line curve) is identical to that of $Y_3Al_2Ga_3O_{12}$: Pr^{3+} phosphor monitored at the Pr^{3+} 307 nm emission (the magenta dot-dash line curve); both consist of one strong broad band peaking at 283 nm and one small band peaking at ~244 nm, which correspond to a transition from the ³H₄ ground state to the lowest 4f5d state of Pr³⁺ (labeled as $4f5d_1$) and a transition from the ${}^{3}H_4$ state to a higher 4f5dstate of Pr³⁺ (labeled as 4f5d₂), respectively.²⁸ After the excitation is ceased, the Y₃Al₂Ga₃O₁₂:Pr³⁺ phosphor exhibits longlasting Pr^{3+} persistent luminescence in the UVB (at ~307 nm), while for the Y₂GdAl₂Ga₃O₁₂:Pr³⁺ phosphor, as expected, the energy released to the Pr³⁺ 4f5d emitting state from the trap-

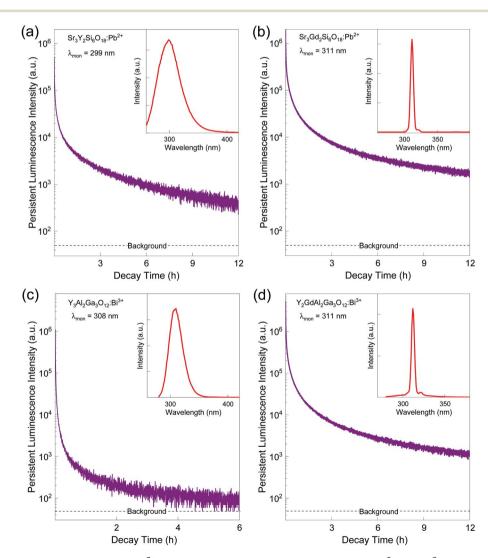


Fig. 4 Persistent luminescence decay curves of Gd^{3+} NB-UVB persistent phosphors developed from Pb^{2+} and Bi^{3+} -activated UVB persistent phosphors. (a) $Sr_3Y_2Si_6O_{18}:Pb^{2+}$ UVB persistent phosphor. (b) $Sr_3Gd_2Si_6O_{18}:Pb^{2+}$ NB-UVB persistent phosphor. (c) $Y_3Al_2Ga_3O_{12}:Bi^{3+}$ UVB persistent phosphor. (d) $Y_2GdAl_2Ga_3O_{12}:Bi^{3+}$ NB-UVB persistent phosphors were pre-irradiated by a 254 nm lamp for 5 min. The inset in each figure shows the persistent luminescence emission spectrum recorded at 10 min decay.

ping states is continuously and efficiently transferred to the $Gd^{3+} {}^{6}P_{J}$ levels, producing Gd^{3+} persistent luminescence in the NB-UVB (Fig. 3b).

In above discussion, we used Pr^{3+}/Gd^{3+} energy transfer ion pair to demonstrate the design of Gd³⁺-activated, Pr³⁺-sensitized NB-UVB persistent phosphors. In addition to Pr³⁺, our design principle is also valid for other sensitizers of Gd³⁺, provided that these sensitizers activated UVC or UVB persistent phosphors are available. As a test, we developed a Pb²⁺-activated UVB persistent phosphor Sr₃Y₂Si₆O₁₈:Pb²⁺, which exhibits a broadband persistent luminescence peaking at 299 nm (attributed to the $Pb^{2+3}P_1 \rightarrow {}^{1}S_0$ emission transition) for longer than 12 h (Fig. 4a). The ${}^{3}P_{1}$ emitting level of Pb²⁺ in the $Sr_3Y_2Si_6O_{18}$:Pb²⁺ phosphor overlaps with the ⁶P₁ levels of Gd³⁺; therefore, when the Y^{3+} ions in the $Sr_3Y_2Si_6O_{18}$:Pb²⁺ phosphor are substituted by Gd³⁺ ions, the resulting Sr₃Gd₂Si₆O₁₈:Pb²⁺ material (see crystal structure in Fig. S1a, ESI[†]) becomes an excellent Gd³⁺ NB-UVB persistent phosphor with an afterglow time of more than 12 h (Fig. 4b). Another material is the Bi^{3+} sensitized Y2GdAl2Ga3O12:Bi3+ NB-UVB persistent phosphor (see crystal structure in Fig. S1b, ESI[†]) developed from a Bi³⁺activated UVB persistent phosphor Y₃Al₂Ga₃O₁₂:Bi³⁺. After charged by a 254 nm lamp, the Y₃Al₂Ga₃O₁₂:Bi³⁺ phosphor emits Bi3+ broadband persistent luminescence peaking at ~308 nm (attributed to the Bi^{3+ 3}P₁ \rightarrow ¹S₀ emission transition) (Fig. 4c), while the Y₂GdAl₂Ga₃O₁₂:Bi³⁺ phosphor exhibits longtime Gd³⁺ 311 nm persistent luminescence due to the efficient persistent energy transfer from the ³P₁ emitting level of Bi³⁺ to the ${}^{6}P_{I}$ levels of Gd³⁺ (Fig. 4d).

Above we have demonstrated the design and fabrication of eight Gd³⁺ NB-UVB persistent phosphors. It is worth noting that our extensive synthesis work reveals that the Gd³⁺ content in the above-mentioned NB-UVB persistent phosphors can be tuned over a great range. For instance, the general chemical formula of the Gd³⁺-containing NB-UVB persistent phosphors developed from the Sr₃Y₂Si₆O₁₈:Pb²⁺ UVB persistent phosphor can be written as $Sr_3Y_{2-x}Gd_xSi_6O_{18}$:Pb²⁺, where x can vary from 0.1 to 2.0 (when x = 2.0, the resulting material is the Sr₃Gd₂Si₆O₁₈:Pb²⁺ phosphor in Fig. 4b), and these Gd³⁺ persistent phosphors exhibit comparable persistent luminescence performance (Fig. S3, ESI[†]). Moreover, our synthesis work also showed that some ions in some hosts can be substituted by other ions of the same group. For instance, the Y³⁺ ions in the Y₃Al₂Ga₃O₁₂ host can be substituted by Lu³⁺ ions to form Lu₃Al₂Ga₃O₁₂ host; therefore, more UVB and NB-UVB persistent phosphors can be fabricated such as the UVB persistent Lu₃Al₂Ga₃O₁₂:Pr³⁺ S4a, ESI†) phosphors (Fig. and Lu₃Al₂Ga₃O₁₂:Bi³⁺ (Fig. S5a, ESI[†]) and the NB-UVB persistent phosphors Lu₂GdAl₂Ga₃O₁₂:Pr³⁺ (Fig. S4b, ESI[†]) and Lu₂GdAl₂Ga₃O₁₂:Bi³⁺ (Fig. S5b, ESI[†]). Finally, our design concept is expected to lead to the discovery of more Gd³⁺ NB-UVB persistent phosphors sensitized by not only Pr^{3+} , Pb^{2+} and Bi³⁺, but also Ce³⁺ (note: we have not identified Ce³⁺-activated UVC or UVB persistent phosphors; moreover, we have not found the hosts for the Pb2+ and Bi3+ UVC persistent luminescence).

Conclusions

We have proposed and tested a design concept to develop Gd³⁺-activated NB-UVB persistent phosphors from the Pr³⁺-, Pb²⁺- or Bi³⁺-activated UVC or UVB persistent phosphors. Guided by this design concept, we have fabricated ten Gd^{3+} NB-UVB persistent phosphors including Pr³⁺-sensitized Sr₃Gd₂Si₆O₁₈:Pr³⁺, Ca_{1.8}Gd_{0.2}Al₂SiO₇:Pr³⁺, Lu_{1.8}Gd_{0.2}SiO₅:Pr³⁺, $Ca_{2.8}Gd_{0.2}Al_{2}Si_{3}O_{12}:Pr^{3+},\ LiY_{0.8}Gd_{0.2}SiO_{4}:Pr^{3+},\ Y_{2}GdAl_{2}Ga_{3}O_{12}:Pr^{3+},\ Y_{2}GdAl_{2}Ga_{3}O$ Pr^{3+} and $Lu_2GdAl_2Ga_3O_{12}$: Pr^{3+} , Pb^{2+} -sensitized $Sr_3Gd_2Si_6O_{18}$: Pb^{2+} , and Bi^{3+} -sensitized $Y_2GdAl_2Ga_3O_{12}:Bi^{3+}$ and Lu₂GdAl₂Ga₃O₁₂:Bi³⁺. Moreover, we have also developed five first-ever Pr³⁺-, Pb²⁺- and Bi³⁺-activated persistent phosphors emitting in the UVB region including Y₃Al₂Ga₃O₁₂:Pr³⁺, $Lu_3Al_2Ga_3O_{12}$: Pr^{3+} , $Sr_3Y_2Si_6O_{18}$: Pb^{2+} , $Y_3Al_2Ga_3O_{12}$: Bi^{3+} and $Lu_3Al_2Ga_3O_{12}$:Bi³⁺. Therefore, this study expands the persistent luminescence research to the NB-UVB as well as the broader UVB spectral regions. The realization of Gd³⁺ NB-UVB persistent luminescence can potentially revolutionize the way the NB-UVB light is used. For instance, the Gd³⁺ NB-UVB persistent phosphors may be used as self-sustained glowing NB-UVB radiation sources for phototherapy, as illustrated by the conceptual NB-UVB Band-Aid phototherapy in Fig. S6 (ESI†).

Author contributions

Zhengwei Pan: conceptualization, methodology, funding acquisition, supervision, writing-reviewing and editing. Peter Kner: funding acquisition, supervision, writing-reviewing and editing. Xianli Wang: investigation, data curation, writing – original draft. Yafei Chen: investigation, data curation, writingreviewing and editing.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work is supported by the US National Science Foundation (DMR-1705707) and the College of Petroleum Engineering and Geosciences, King Fahd University of Petroleum & Minerals. We thank McPherson, Inc. (Chelmsford, MA, USA) for helping measure the photoluminescence excitation and emission spectra.

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