

**Carbon Dioxide and Hydrogen Sulfide Removal from Simulated Landfill Gas Using Steel  
Slag**

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Manuscript Submitted to:

*Journal of Environmental Engineering, ASCE*

July 25, 2020

## Abstract

Municipal solid waste landfills are a source of major greenhouse gases such as methane ( $\text{CH}_4$ ) and carbon dioxide ( $\text{CO}_2$ ) while also emit trace amount of hydrogen sulfide ( $\text{H}_2\text{S}$ ). Recently, steel slag has extensively been used for mineral  $\text{CO}_2$  sequestration to minimize the  $\text{CO}_2$  releases to the atmosphere. This study explores the potential of basic oxygen furnace slag (BOF) steel slag to simultaneously remove  $\text{CO}_2$  and  $\text{H}_2\text{S}$  from landfill gas (LFG). Various batch and column tests were conducted to evaluate the  $\text{CO}_2$  and  $\text{H}_2\text{S}$  removal potential of the BOF slag under various conditions such as moisture content and particle size of the BOF slag. The three different particle sizes of BOF slag (coarse, as-is, and fine) were exposed to continuous flow of a synthetic LFG [50%  $\text{CO}_2$ , 48.25%  $\text{CH}_4$  and 1.75%  $\text{H}_2\text{S}$  by volume ( $v/v$ )] in a column reactor to evaluate the effect of particle size on  $\text{CO}_2$  and  $\text{H}_2\text{S}$  removal capacity of the slag. Similarly, the BOF slag was exposed to synthetic LFG as well as 20% ( $v/v$ ) of  $\text{H}_2\text{S}$  alone in batch reactors at varying moisture contents (10–30% by weight) to evaluate the effect of moisture content on the  $\text{CO}_2$  and  $\text{H}_2\text{S}$  removal capacity of the slag. A significant  $\text{H}_2\text{S}$  removal of 27 g  $\text{H}_2\text{S}$   $\text{kg}^{-1}$  BOF slag and  $\text{CO}_2$  removal of 76 g  $\text{CO}_2$   $\text{kg}^{-1}$  BOF slag were obtained in the batch reactor. The fine BOF slag ( $< 0.106$  mm) showed the maximum  $\text{CO}_2$  removal (300 g  $\text{CO}_2$   $\text{kg}^{-1}$  BOF slag) and  $\text{H}_2\text{S}$  removal (38 g  $\text{H}_2\text{S}$   $\text{kg}^{-1}$  BOF slag) upon exposure to continuous synthetic LFG flow in the column reactor. The quantitative X-Ray diffraction (QXRD) analysis showed highest increase in carbon (77.5 g C  $\text{kg}^{-1}$  BOF slag) and sulfur (28 g S  $\text{kg}^{-1}$  BOF slag) contents in the fine BOF slag which was consistent with the mass balance of carbon and sulfur from  $\text{CO}_2$  and  $\text{H}_2\text{S}$  uptake in column tests. The major reaction product with  $\text{H}_2\text{S}$  was elemental sulfur depicted by the significant increase in the sulfur content in the X-ray fluorescence analysis. The key minerals involved in carbonation reactions were lime, portlandite, and larnite as these minerals showed significant reduction in weight percentage (100%, 82% and

45 80%, respectively) in the QXRD analysis. Overall, BOF slag showed promising results in  
46 mitigating CO<sub>2</sub> and H<sub>2</sub>S from LFG.

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48 **Keywords:** Municipal solid waste; landfill gas emissions; BOF slag; carbon dioxide removal;  
49 hydrogen sulfide removal.

50

## Introduction

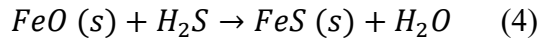
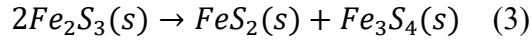
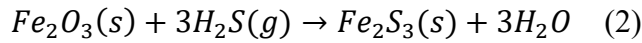
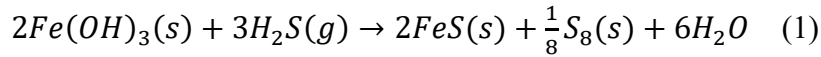
Municipal solid waste (MSW) landfills are the third largest source of anthropogenic methane ( $\text{CH}_4$ ) emissions in the United States and are also a major source of carbon dioxide ( $\text{CO}_2$ ) emissions, each constituting 50% of the landfill gas (LFG) composition (USEPA, 2020), and odor problems (Ng et al. 2017; Lee et al. 2017).  $\text{CH}_4$  and  $\text{CO}_2$  are two of the most abundant greenhouse gases and principal contributors to the global climate change. Odor control persists as one of the key concerns for landfill operators due to the malodorous compounds such as hydrogen sulfide ( $\text{H}_2\text{S}$ ), methyl mercaptan, and methyl sulfide (He et al. 2011). For example, LFG from MSW landfills contains 100 to 1,000 ppm  $\text{H}_2\text{S}$  (Ng et al. 2017) which is a major odor causing component, even at very low concentrations [0.01 to 1.5 parts per million (ppm)] (Ko et al. 2015).  $\text{H}_2\text{S}$  concentrations above 100 ppm are immediately dangerous to life and health as stated by the U.S. Occupational Safety and Health Administration and the permissible exposure limit (8-hour limit) for  $\text{H}_2\text{S}$  is 10 ppm (OSHA, 2020). In addition,  $\text{H}_2\text{S}$  is also highly flammable with explosive range in air at 4.5 to 45.5% by volume (v/v) (OSHA, 2020). Hence,  $\text{H}_2\text{S}$  emissions from landfills is a serious concern.

Active landfills having waste-to-energy plants and/or  $\text{CH}_4$  recovery plants use various technologies to purify LFG to usable  $\text{CH}_4$  products. The LFG purification process involves removal of moisture, particulates, and gases (such  $\text{CO}_2$  and  $\text{H}_2\text{S}$ ), and volatile organic compounds such as siloxanes (Peterson 2013). Various processes to scrub  $\text{CO}_2$  and  $\text{H}_2\text{S}$  from LFG involve physical and chemical adsorption. Adsorption using activated carbon, zinc oxide, and iron oxide and amine scrubbing (such as Selexol) are the most common  $\text{H}_2\text{S}$  removal technologies but are cost intensive (Deed et al. 2004; Spiegel et al. 1995). Similarly, adsorption on activated carbon

beds, silicates or molecular sieves, dissolution of CO<sub>2</sub> in the solvents forming carbonates, water scrubbing, etc. are some of the popular techniques for removing CO<sub>2</sub> from LFG (Schumacher 1983). Although there are many technologies available to treat H<sub>2</sub>S and CO<sub>2</sub> in LFG, they are used mostly during CH<sub>4</sub> recovery process to obtain high purity CH<sub>4</sub> and hence are feasible when the CH<sub>4</sub> concentrations in the LFG are high (Arkharov et al. 2016). Similarly, older landfills do not have gas collection systems, and even at modern engineered landfills with the gas collection systems fugitive LFG emissions persist due to the limited efficiency of gas recovery systems (Scheutz et al. 2009). Accordingly, alternative approaches are needed to address fugitive LFG emissions migrating through landfill capping systems.

Steel slag, a byproduct from steel making and refining processes, are generated in huge quantity every year of which nearly 60–85% are used in construction applications and the remaining 15–40% are either stockpiled or landfilled (Yildirim and Prezzi 2015). Because of its alkalinity, the use of steel making slag has recently been explored for mineral carbon sequestration applications (Huijgen et al. 2005; Bonenfant et al. 2008; Chetri et al. 2019). Steel slag is primarily composed of calcium (Ca), iron (Fe), magnesium (Mg) and silicon (Si) which exist in both crystalline and amorphous phases (Caicedo-Ramirez et al. 2020; Yildirim and Prezzi 2011). Steel slag has shown promising potential to sequester CO<sub>2</sub> in the previous studies (Huijgen et al. 2005; Bonenfant et al. 2008; Ukwattage et al. 2017; Chetri et al. 2019). Iron oxides have been commonly used in sweetening of natural gas or landfill gas (Davydov et al. 1998). Iron oxides serve as suitable oxidizing agents for H<sub>2</sub>S removal (Cantrell et al. 2003; Więckowska 1995). Iron has been found to be the metal of choice for the removal of H<sub>2</sub>S as H<sub>2</sub>S reduces Fe<sup>3+</sup> to Fe<sup>2+</sup> under anerobic condition as shown in **Eq. 1** (Bergersen and Haarstad 2008). The ferric oxide (Fe<sub>2</sub>O<sub>3</sub>) reacts with H<sub>2</sub>S to produce Fe<sub>2</sub>S<sub>3</sub> which is thermodynamically unstable and reacts further to form pyrite (FeS<sub>2</sub>)

and ferric sulfide ( $\text{Fe}_3\text{S}_4$ ) (Davydov et al. 1998) as shown in **Eq. 2** and **3**. In addition, the ferrous oxide ( $\text{FeO}$ ) can react with  $\text{H}_2\text{S}$  in aqueous solution to form ferrous sulfide ( $\text{FeS}$ ) as shown in **Eq. 4** (Sarperi et al. 2014; Rickard and Luther 2007). Similarly, the high alkalinity of BOF slag favors the dissociation of  $\text{H}_2\text{S}$  into  $\text{HS}^-$  (bisulfide) and  $\text{S}^{2-}$  (sulfide) ( $\text{pK}_a$  for  $\text{H}_2\text{S}/\text{HS}^- = 7$ ,  $\text{HS}^-/\text{S}^{2-} > 11$ ) which in turn react with the metal ions such as iron leading to the formation of elemental sulfur and metal sulfides (Montes Moran et al. 2012).



Owing to the high iron oxide content of the steel slags, they have been explored as an alternative for conventional absorbents for removing  $\text{H}_2\text{S}$  from gas streams and sediments as summarized in **Table 1**. For example, Montes-Morán et al. (2012) exposed < 212 mm and 212-500 mm sized Linz-Donawitz (LD) slag to 1,000 ppmv ( $1.39 \text{ g/m}^3$  at  $25^\circ\text{C}$  and 1 atm)  $\text{H}_2\text{S}$  gas at relative humidity (RH) of 10 and 50% in a fixed bed reactor and observed a maximum  $\text{H}_2\text{S}$  removal of 180 g  $\text{H}_2\text{S}$ /kg slag for 212-500 mm sized slag. Kim et al. (2012) and Asaoka et al. (2013) analyzed  $\text{H}_2\text{S}$  removal from the  $\text{H}_2\text{S}$  solution using steel slag by simulating conditions that prevail in pore water of organically enriched sediments. Apart from testing  $\text{H}_2\text{S}$  removal potential of BOF slag in batch reactor (**Table 1**), Sarperi et al. (2014) used BOF slag to treat biogas from anaerobic digester with an aim to upgrade the raw biogas for vehicle fuel use (bio-CNG). The slag showed complete removal of  $\text{H}_2\text{S}$  from raw biogas consisting of 250 ppmv of  $\text{H}_2\text{S}$  in the 2-hour exposure time. Some studies such as Caicedo-Ramrez et al (2020) used basic oxygen furnace (BOF) steel

stag in anaerobic sludge digesters to sequester phosphate and H<sub>2</sub>S and achieved 78% removal of H<sub>2</sub>S from biogas in 160 days of continuous operation.

Although plenty of information exists on purification of biogas using iron oxides as well as CO<sub>2</sub> sequestration with steel slag, little is known about the simultaneous interaction of CO<sub>2</sub> and H<sub>2</sub>S with the steel slag and the underlying reaction mechanisms. Most of the studies focus on removal of CO<sub>2</sub> and H<sub>2</sub>S by steel slag independently and none of the studies have explored the potential of BOF slag to remove CO<sub>2</sub> and H<sub>2</sub>S simultaneously under landfill conditions. The main objectives of this study are to investigate the potential of BOF slag to remove CO<sub>2</sub> and H<sub>2</sub>S from the LFG and to identify the basic mechanisms responsible for the removal process. In this regard, laboratory batch tests and column tests were conducted to quantify the simultaneous CO<sub>2</sub> and H<sub>2</sub>S removal capacity of the slag and analyze the effect of various parameters such as moisture content and slag particle size on the CO<sub>2</sub> and H<sub>2</sub>S removal capacity.

## **Materials and Methods**

### ***Material Preparation and Characterization***

The BOF slag was obtained from Arcelor Mittal Steel Plant, Indiana Harbor East, Indiana, USA. Freshly crushed and dried BOF slag sample was obtained with a top sieve size of 10 mm (3/8 in.) The samples were stored in 5-gallon buckets with airtight lids to prevent carbonation reactions with atmospheric CO<sub>2</sub>. The specific gravity of the slag was determined as per ASTM D854-14 (2014). The loss-on-ignition (LOI), which typically represents organic content, was determined as per ASTM D2974 (2020) and particle size distribution was evaluated as per ASTM D6913/6913M

(2017) and D7928 (2017). The pH and redox (oxidation reduction) potential of the slag was measured using a pH meter (Orion 720A model, Orion Research, Inc. Beverly, Ma, USA) at a liquid to solid ratio of 4:1. Water holding capacity (WHC) of slag was determined by measuring the water retained in the slag after gravity drainage.

Three different gradations of BOF slag were used. The BOF slag obtained from the plant in its original particle size distribution was termed as “as-is”. BOF slag passing US No. 4 sieve (4.75 mm) and retained on US No. 60 (0.25 mm) was termed as “coarse” and the BOF slag passing US No. 140 sieve (0.106 mm) was termed as “fine” slag. The metal oxides and minerals present in the BOF slag were quantified before and after experimentations using X-ray fluorescence (XRF) and quantitative X-ray powder diffraction (QXRD) with Rietveld quantification analysis using procedures outlined by Reddy et al. (2019a). The morphological changes in the BOF slag were analyzed by scanning electron microscopy (SEM) using a JEOL JSM-6320F High Resolution Scanning Microscope operated at 2.5 kV accelerating voltage and Oxford X-Ray Energy Dispersive Spectrometer (XEDS) fitted with a Hitachi S-3000N Variable Pressure Electron Microscope, respectively.

### ***Batch Experiments***

Two suites of batch experiments were conducted along with the controls (without slag). Suite I was conducted with the fresh as-is BOF slag at four different moisture contents [MCs; 0, 10, 20, 30% by weight ( $w/w$ )]. The BOF slag (1 g) was placed in a 125 ml glass serum vial (WHEATON®, Millville, NJ, USA) and required amount of deionized water was added to each vial to achieve the target MC. The vial was then purged with synthetic LFG [50% CO<sub>2</sub>, 48.25% CH<sub>4</sub> and 1.75% H<sub>2</sub>S



(v/v)] and hermetically sealed with rubber septa and secured with aluminum crimp cap. The vials were shaken rigorously for approximately 30 seconds to facilitate contact between the slag and the LFG. The headspace gas was analyzed at regular intervals until 24 hours. Suite II was conducted following similar procedure except that the slag was exposed to 20% H<sub>2</sub>S (v/v) and the MCs tested were 10, 15 and 20% (w/w), and the headspace gas was analyzed until the concentration of H<sub>2</sub>S became invariant.

For both suites, samples were extracted from the vials with a 1 mL gas tight syringe and were analyzed using SRI 9300 gas chromatography (GC) machine (SRI Instruments, Torrance, CA, USA) equipped with thermal conductivity detector (TCD) for the detection of CH<sub>4</sub> and CO<sub>2</sub>. SRI Model 110 stand-alone detector chassis with electronic pressure control equipped with flame ionization/flame photometric detector (FID/FPD) was connected to the host GC with a heated transfer line for the detection of H<sub>2</sub>S. HayeSep-D packed column 6' x 1/8'' for separation of CH<sub>4</sub> and CO<sub>2</sub>, and MXT-1 capillary column for separation of H<sub>2</sub>S were fitted in the host GC. A T-connection was attached to the inlet port of the host GC for splitting of the gas samples into the two columns. Helium and hydrogen were used as the carrier gases for TCD and FID/FPD, respectively.

### ***Column Experiments***

Column experiments were conducted to gain insights on the interaction of CO<sub>2</sub> and H<sub>2</sub>S with the slag under continuous LFG flow condition and the effect of slag particle size on the CO<sub>2</sub> and H<sub>2</sub>S removal. Experiments were conducted in Chromaflex glass columns (Cole-Parmer, Vernon Hills, Illinois, USA) of 30 cm height and 2.5 cm internal diameter, top and bottom outfitted with bed

support meshes, end connections, and screw caps, as detailed by Reddy et al. (2019a, b). Three BOF slag gradations (as-is, coarse, and fine) at a MC of 10% (w/w) were packed using 5 cm lifts and light tamping in each column. Dry synthetic LFG [50% CO<sub>2</sub>, 48.25% CH<sub>4</sub> and 1.75% H<sub>2</sub>S (v/v)] was injected into each column in an upward flow mode at an inlet pressure of ~10 psi (69 kPa). Dry gas was used to avoid the potential for immediate precipitation of sulfur from interaction of H<sub>2</sub>S with water during humidification. The inlet flow rate was controlled by flowmeters (Cole-Parmer, Model No. PMRI-010874). The inlet flowrates of 6-12 ml/min were maintained to ensure the following fluxes; 1) CH<sub>4</sub>: 5,500-10,500 g day<sup>-1</sup> m<sup>-2</sup>; 2) CO<sub>2</sub>: 16,000-30,560 g day<sup>-1</sup> m<sup>-2</sup>; and 3) H<sub>2</sub>S: 430-830 g day<sup>-1</sup> m<sup>-2</sup>. These fluxes used were much higher than the field reported CH<sub>4</sub> fluxes ranging from 0.0002 to 4,000 g day<sup>-1</sup> m<sup>-2</sup> (De Visscher et al. 1999). The inlet and outlet gas samples were analyzed regularly with the GC. Two sets of column experiments were performed:

1) In the first set, two column tests were performed with as-is BOF slag. One test was terminated just after the breakthrough of CO<sub>2</sub> gas, whereas the second when CO<sub>2</sub> and H<sub>2</sub>S removal capacity was exhausted (e.g. C<sub>out</sub>/C<sub>in</sub> of CO<sub>2</sub> and H<sub>2</sub>S  $\cong$  1). The main goal was to quantify the major mineralogical changes associated with short-term and long-term CO<sub>2</sub> and H<sub>2</sub>S removal.

2) In the second set, three column tests were performed with coarse, as-is and fine BOF slag each. The tests were terminated when the CO<sub>2</sub> and H<sub>2</sub>S removal capacity was exhausted (C<sub>out</sub>/C<sub>in</sub> of CO<sub>2</sub> and H<sub>2</sub>S  $\cong$  1). Here the goal was to investigate the effect of BOF slag particle size on removal capacity and identify the principle reaction mechanisms for the removal of CO<sub>2</sub> and H<sub>2</sub>S by BOF slag.

XRF and QXRD analyses were conducted on the BOF slag samples at several key intervals of interest to identify the elemental and mineralogical changes in the slag due to reaction with CO<sub>2</sub>

and H<sub>2</sub>S. Additionally, SEM-EDS analyses were performed on the slag samples before and after exposure to synthetic LFG and 20% H<sub>2</sub>S (v/v) to evaluate the morphological changes in the slag.

## **Results and Discussion**

### ***Material Characterization***

Specific gravity and pH of the BOF slag were 3.34 and 12.4, respectively, which are consistent with the previous studies (Reddy et al. 2019c). **Figure 1** shows the particle size distribution of each BOF slag sample. The coarse BOF slag was classified as poorly graded sand (SP) per Unified Soil Classification System (USCS) with mean particle size (D<sub>50</sub>) of 1.8 mm. The as-is BOF slag had fines content of ~ 14%, D<sub>50</sub> of 0.82 mm, and classified as silty sand (SM). The fine BOF slag had D<sub>50</sub> of 0.06 mm and classified as low plasticity silt (ML). The LOI/organic content of the slag was low (1.94%) which suggested lack of carbonation for the initial condition. The WHC capacity of the slag was 30% (w/w) which is comparable to the landfill cover soils (Reddy et al. 2019a).

**Tables 2, 3 and 4** summarize the elemental composition, bulk chemistry, and mineral phases, respectively, present in the BOF slag as a function of various exposure conditions. The major elements present in the slag are Ca, Fe, Mg, and Si (**Table 2**). The ranges of metal oxides in the original (fresh) BOF slag (**Table 3**) are consistent with the bulk chemistry reported in other studies (Yildirim and Prezzi, 2011; Reddy et al. 2019c; Shi 2004). The original (fresh) BOF slag contained minerals (**Table 4**) known to have high affinity for CO<sub>2</sub> and H<sub>2</sub>S (Huijgen et al. 2006; Reddy et al. 2019a, b; Ng et al. 2017; Kim et al. 2012; Montes-Morán et al. 2012). Free lime (CaO), portlandite [Ca(OH)<sub>2</sub>] and larnite (Ca<sub>2</sub>SiO<sub>4</sub>) have the potential to bind CO<sub>2</sub> as stable carbonates,

as detailed elsewhere (Reddy et al. 2019a, b). The slag contains high amount of  $\text{Fe}_2\text{O}_3$  (29.7%) and  $\text{FeO}$  (9.6%) which are known to react with  $\text{H}_2\text{S}$  as shown in **Eqs. 2 and 4** (Montes-Morán et al. 2012; Kim et al. 2012; Cantrell et al. 2003; Davydov et al. 1998; Sarperi et al. 2014).

### ***Batch Experiments***

Batch experiments were performed to analyze the interaction of BOF slag with synthetic LFG components ( $\text{CH}_4$ ,  $\text{CO}_2$  and  $\text{H}_2\text{S}$ ). There was no noticeable absorption of  $\text{CO}_2$  at 0% (w/w) MC (in dry state) of the BOF slag as shown in **Figure 2**.  $\text{CO}_2$  removal increased sharply within one hour of exposure to LFG, and the  $\text{CO}_2$  concentration reduced from 50% (v/v) to nearly 15% (v/v) leading to 70% removal of  $\text{CO}_2$  when the moisture was increased from 0 to 10% (w/w) (**Figure 2a**). However, the  $\text{CO}_2$  removal rate did not vary significantly when the moisture was increased from 10 to 30% as depicted by the similar slopes of  $\text{CO}_2$  concentration versus time plot in **Figure 2a**. The  $\text{CO}_2$  removal capacity of BOF slag reduced gradually after one hour as depicted by the change in slope of  $\text{CO}_2$  concentration curve from nearly 20 to 2 in the first 3 hours and by 5 hour the slope was essentially asymptotic (**Figure 2a**). An overall  $\text{CO}_2$  removal of 68-76 g  $\text{CO}_2$   $\text{kg}^{-1}$  BOF slag was obtained which is consistent with the  $\text{CO}_2$  sequestration capacity reported in Reddy et al. (2019a). Consistent with  $\text{CO}_2$ , there was no  $\text{CH}_4$  removal in the dry state (0% MC) (**Figure 2b**). Under moist conditions, the slag showed nominal removal of  $\text{CH}_4$ , leading to nearly 20–26% removal or 8–11 g  $\text{CH}_4$   $\text{kg}^{-1}$  BOF slag. Similar to  $\text{CO}_2$  and  $\text{CH}_4$ , the slag showed no affinity towards  $\text{H}_2\text{S}$  in dry state (**Figure 2c**). However,  $\text{H}_2\text{S}$  removal was significantly affected by moisture leading to 100% removal within 24 hours for all MCs tested (**Figure 2c**). There was no significant impact of presence of  $\text{H}_2\text{S}$  on the  $\text{CO}_2$  removal capacity or vice versa. **Figure 2d** shows the corresponding

H<sub>2</sub>S removal capacity as g H<sub>2</sub>S kg<sup>-1</sup> BOF slag. Since the initial concentration of H<sub>2</sub>S in the synthetic LFG was low (~1.75%), a removal of only 2.7 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag was obtained when the amount of H<sub>2</sub>S in the batch reactor was completely exhausted.

The H<sub>2</sub>S with initial concentration of 1.75% (v/v) was rapidly removed, as a result additional series of batch experiments were performed with higher initial H<sub>2</sub>S concentration of 20% (v/v) at various MCs to evaluate the maximum H<sub>2</sub>S removal capacity of the BOF slag (**Figure 3**). **Figure 3a** shows H<sub>2</sub>S concentrations as a function of exposure time and initial MC. An increasing H<sub>2</sub>S removal was observed with MC. After one hour, the BOF slag with 20% (w/w) MC showed more cumulative H<sub>2</sub>S removal than at lower MC values (**Figure 3b**), however, the removal capacity dropped off significantly after an hour. After 120 hours, the H<sub>2</sub>S removal capacity was nearly saturated (**Figure 3a**) attaining 17.8, 21.7 and 27 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag, respectively as a function of time (**Figure 3b**).

One of the most important reaction steps for carbonation of steel slag is dissolution of minerals and CO<sub>2</sub> gas in the solution resulting in the precipitation of carbonates (Librandi et al 2019; Ukwattage et al. 2017; O'Connor et al. 2001). Hence, moisture plays an important role but increasing the MC above 10% (w/w) does not significantly improve/impact the carbonation capacity of the BOF slag. The steep slope of the CO<sub>2</sub> removal plot in **Figure 2a** likely corresponds to the free CaO, Ca(OH)<sub>2</sub> and Ca<sub>2</sub>SiO<sub>4</sub> of the slag (**Table 4**, fresh as-is) readily reacting with CO<sub>2</sub> as these minerals showed sharp decrease in weight % at CO<sub>2</sub> breakthrough (in column tests) which is discussed in detail in the next section. The gradual decrease in the slope of the CO<sub>2</sub> removal curve suggests progressive dissolution of the minerals from the inner core of the BOF slag. The BOF slag showed some noticeable removal of CH<sub>4</sub> (**Figure 2b**) which could be mere adsorption. **Figure 2c** shows the influence of MC on H<sub>2</sub>S removal was stronger than that for CO<sub>2</sub> removal.

Sarperi et al. (2014) reported dramatic differences in H<sub>2</sub>S removal as a function of moisture, 37 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag at 0% (w/w) MC versus 142 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag at 35% (w/w) MC. The BOF slag showed significantly higher H<sub>2</sub>S removal rates (0.53 g H<sub>2</sub>S hr<sup>-1</sup>) at 30% (w/w) MC than at lower MCs (0.11–0.13 g H<sub>2</sub>S hr<sup>-1</sup>) (**Figure 2d**) which is consistent with the previous findings. The increase in H<sub>2</sub>S removal was not very significant when the MC was increased from 10 to 20% (w/w), but rapid removal of H<sub>2</sub>S occurred when the MC increased from 20 to 30% (w/w). The long-term exposure of BOF slag to 20% (v/v) H<sub>2</sub>S showed an increasing trend in H<sub>2</sub>S removal with moisture (**Figure 3**) consistent with the observations of Sarperi et al. (2014) whose H<sub>2</sub>S removal was 37, 119 and 142 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag at 0, 20 and 35% (w/w) MCs, respectively. Sarperi et al. (2014) reported very high H<sub>2</sub>S removal capacity (142 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag) for continuous stirring of batch reactors at 300 rpm. The slag with 20% (w/w) MC showed significantly higher H<sub>2</sub>S removal (2.5–4 times) in first one hour than that with 10% and 15% (w/w) MCs (**Fig. 3b**) which could be attributed to the rapid dissolution of the H<sub>2</sub>S in the presence of higher moisture leading to significant sulfur precipitation (**Eq. 1**). Similarly, the high pH of the slag favors the absorption of the H<sub>2</sub>S as the pK<sub>a</sub> for H<sub>2</sub>S/HS<sup>-</sup> is 7. Moisture also plays an important role in the reductive dissolution of iron oxides by H<sub>2</sub>S. Ferric oxide (Fe<sub>2</sub>O<sub>3</sub>) precipitates as ferric hydroxide [Fe(OH)<sub>3</sub>] in solution. Davydov et al. (1998) showed that the amorphous Fe(OH)<sub>3</sub> expresses more reactivity with H<sub>2</sub>S than the same mass of anhydrous Fe<sub>2</sub>O<sub>3</sub>. **Figure 3a** shows two stage reaction mechanism for the H<sub>2</sub>S removal, initial rapid removal with nearly 70–90% of total removal occurring in the first 24 hours which was followed by a relatively slower removal rate (flatter H<sub>2</sub>S removal curves after 24 hours). The initial rapid removal could be attributed to the interaction of H<sub>2</sub>S with the surficial Fe(OH)<sub>3</sub> and formation of elemental sulfur as shown in **Eq. 1**. The second stage with relatively slower H<sub>2</sub>S removal rate could be attributed to reductive dissolution of Fe<sub>2</sub>O<sub>3</sub> lying

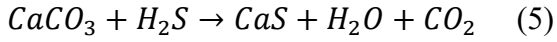
below the surface layers forming bulk ferric sulfides ( $\text{Fe}_2\text{S}_3$ ) as shown in **Eq. 2** (Davydov et al. 1998).

### ***Column Experiments***

The first set of column experiments were conducted with as-is BOF slag. One experiment was stopped shortly after breakthrough of  $\text{CO}_2$  (~100 PV) and another was continually exposed to synthetic LFG until  $C_{\text{out}}/C_{\text{in}}$  of  $\text{CO}_2$  and  $\text{H}_2\text{S}$  became invariant. **Figure 4** shows the breakthrough curve for as-is BOF slag at short term (breakthrough) and long term ( $C_{\text{out}}/C_{\text{in}}$  of  $\text{CO}_2$  and  $\text{H}_2\text{S} \cong 1$ ) exposure conditions. The second set of column tests was conducted with coarse and fine BOF slag under similar conditions as the first set and was continued until  $C_{\text{out}}/C_{\text{in}}$  of  $\text{CO}_2$  and  $\text{H}_2\text{S}$  became invariant. **Figures 5a** and **5b** show  $\text{CO}_2$  and  $\text{H}_2\text{S}$  removal potential of coarse, as-is and fine BOF slag in terms of variation of  $C_{\text{out}}/C_{\text{in}}$  with inflow PV of the synthetic LFG. The breakthrough of  $\text{CO}_2$  happened at around 40, 80 and 200 PVs for coarse, as-is and fine BOF slags, respectively (**Figure 5a**). Similarly, the breakthrough of  $\text{H}_2\text{S}$  occurred at 100, 200 and 2000 PVs for coarse, as-is and fine BOF slags, respectively (**Figure 5b**).

**Figures 6a** and **6b** show cumulative removal of  $\text{CO}_2$  and  $\text{H}_2\text{S}$ , respectively in terms of inflow PVs of synthetic LFG. The  $\text{CO}_2$  removal capacity of BOF slag was mobilized earlier than the  $\text{H}_2\text{S}$  removal capacity which could be attributed to the pH dependency of the carbonation reactions. For example, the pH of the fine slag dropped to 9.88 from the initial pH of 12.3 and since the  $\text{pK}_a$  of  $\text{HCO}_3^-/\text{CO}_3^{2-}$  is 10.3, the reduction in pH below 10.3 would limit the availability of  $\text{CO}_3^{2-}$  ions for binding Ca ions. On the other hand,  $\text{pK}_a$  of  $\text{H}_2\text{S}/\text{HS}^-$  is 7 which shows higher availability of  $\text{HS}^-$  ions in the system leading to continued absorption of  $\text{H}_2\text{S}$ . The prolonged absorption of  $\text{H}_2\text{S}$  also

suggests possibility of reaction of H<sub>2</sub>S with calcium carbonate (CaCO<sub>3</sub>) as shown in **Eq. 5** (Lin et al. 1995).



The CO<sub>2</sub> removal capacity of slag exhausted ( $C_{out}/C_{in} \sim 1$ ) at around 1,500 PVs of inflow LFG for all three particle sizes of the BOF slag whereas H<sub>2</sub>S removal capacity exhausted at different inflow PVs for different particle sizes of the slag. For example,  $C_{out}/C_{in}$  of H<sub>2</sub>S reached  $\sim 1$  at 1,200, 1,800 and 3,600 PVs for coarse, as-is and fine BOF slag, respectively. The fine BOF slag showed highest cumulative CO<sub>2</sub> removal ( $\sim 300$  g CO<sub>2</sub> kg<sup>-1</sup> BOF slag) followed by as-is ( $\sim 120$  g CO<sub>2</sub> kg<sup>-1</sup> BOF slag) and coarse ( $\sim 80$  g CO<sub>2</sub> kg<sup>-1</sup> BOF slag) BOF slag which is consistent with the observations of Reddy et al. (2019b). The highest H<sub>2</sub>S removal was obtained for fine BOF slag which was  $\sim 38$  g H<sub>2</sub>S kg<sup>-1</sup> BOF slag (**Figure 6b**). The H<sub>2</sub>S removal capacity of as-is and coarse BOF slags was mobilized much earlier than the fine BOF slag resulting in a H<sub>2</sub>S removal of  $\sim 4$  g H<sub>2</sub>S kg<sup>-1</sup> BOF slag and  $\sim 2$  g H<sub>2</sub>S kg<sup>-1</sup> BOF slag, respectively. H<sub>2</sub>S removed at breakthrough by fine BOF slag was 23 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag at 2,000 PVs meaning the slag was able to absorb inflowing H<sub>2</sub>S completely until 2,000 PVs of inflow LFG and it accounts for 60% of the total H<sub>2</sub>S removed by the fine BOF slag. With this H<sub>2</sub>S removal potential, a slag cover of 6 inches thick can sequester H<sub>2</sub>S for nearly 8 years under a continuous H<sub>2</sub>S flux of 3.6 g H<sub>2</sub>S m<sup>-2</sup> d<sup>-2</sup> (flux derived from 85 g CH<sub>4</sub> m<sup>-2</sup> d<sup>-2</sup> reported by Scheutz et al. (2009) assuming 1% H<sub>2</sub>S present in the LFG).

The long-term  $C_{out}/C_{in}$  plot of CO<sub>2</sub> is completely coincident with the short-term plot (breakthrough) (**Figure 4**) which shows complete replication of the experiment and is attributed to the consistency in the dissolution kinetics of the minerals involved in CO<sub>2</sub> removal process. The CO<sub>2</sub> removal potential of the fine slag was significantly higher than the as-is and coarse BOF slag (**Figure 5a**) which was also established in the study by Reddy et al. (2019b). **Figure 5b** shows a



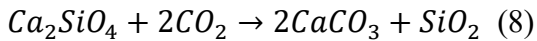
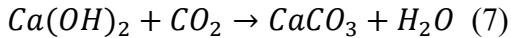
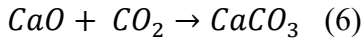
similar pattern in H<sub>2</sub>S removal as CO<sub>2</sub> with fine slag having significantly higher H<sub>2</sub>S removal potential than coarse and as-is. It is interesting to note that the H<sub>2</sub>S removal potential of the BOF slag increased significantly when the particle size changed from as-is to fine whereas, the H<sub>2</sub>S removal potential of as-is and coarse BOF slag were not significantly different. It shows that the higher surface area of the slag fines is creating favorable environment for the adsorption of H<sub>2</sub>S on the slag surface making it available for further reaction with iron oxides. However, in the study by Montes-Moran et al. (2012), the Linz-Donawitz (LD) slag (also known as BOF slag) showed higher H<sub>2</sub>S removal for coarser particle size (212–500 µm) than the finer particle size (< 212 µm) and this behavior was attributed to the higher Fe<sub>2</sub>O<sub>3</sub> content in the larger particle size fractions. In this study, the high surface area of the fine particles seems to have outweighed the higher Fe<sub>2</sub>O<sub>3</sub> content in the coarse and as-is BOF slag in the H<sub>2</sub>S removal process.

#### ***Mineralogical Changes in Slag Exposed to LFG***

**Tables 2** shows the elemental changes in the as-is (short term and long term) and fine BOF slag exposed to synthetic LFG in column reactor sampled from the bottom portion (close to LFG inlet) of the column. The fine slag showed significant increase in the carbon content, followed by as-is (long-term) and as-is (short-term) (**Table 2**). The increase in carbon content in fine slag corresponds to 77.5 g C kg<sup>-1</sup> BOF slag which is consistent with CO<sub>2</sub> removal of the fine slag in column reactor (81.8 g C kg<sup>-1</sup> BOF slag or 300 g CO<sub>2</sub> kg<sup>-1</sup> BOF slag). Similarly, the increase in sulfur content in fine slag corresponds to 28 g S kg<sup>-1</sup> BOF slag which is consistent with the H<sub>2</sub>S removal estimated from the column reactor (35 g S kg<sup>-1</sup> BOF slag or 38 g H<sub>2</sub>S kg<sup>-1</sup> BOF slag).

Calcium oxide (CaO) was reduced from 38.5 wt% to 36.6 wt% in as-is BOF slag just after breakthrough and 34.9% at the end of the experiment (after complete mobilization of carbonation capacity) (**Table 3**). The CaO content in fine BOF slag after long term exposure to LFG was 31.3% which is significantly lower than the fresh BOF slag (**Table 3**). A significant increase in LOI was also observed in the fine BOF slag which indicates deposition of carbonates formed due to carbonation. There was 40% reduction in the free CaO content at the breakthrough and 43% reduction at the end of the experiment in as-is BOF slag which shows most of the free CaO reacts until breakthrough. The fine BOF slag showed 100% reduction of free CaO content after carbonation (**Table 4**) which was attributed to the higher surface area leading to higher availability of free CaO for reaction with CO<sub>2</sub>. Apart from free CaO, amount of Ca(OH)<sub>2</sub> and Ca<sub>2</sub>SiO<sub>4</sub> also decreased following carbonation of the slag. Ca(OH)<sub>2</sub> showed 60.2% decrease at breakthrough and 78.6% at the end of the experiment in as-is BOF slag. It shows that most of the Ca(OH)<sub>2</sub> reacted in the initial phase of the reaction (until breakthrough) after which the decrease in Ca(OH)<sub>2</sub> was gradual leading to only additional 18.4% reduction in Ca(OH)<sub>2</sub> after breakthrough to the termination of the experiment. The fine BOF slag showed a reduction of 81.8% Ca(OH)<sub>2</sub> at the end of the experiment which is nearly same as that of as-is BOF slag. The reduced reactivity of Ca(OH)<sub>2</sub> after breakthrough could be due to the deposition of calcium carbonate (CaCO<sub>3</sub>) on the slag surface limiting further interaction of CO<sub>2</sub> with the minerals underlying the surface which was confirmed by the presence of formations resembling to CaCO<sub>3</sub> crystals in the SEM analysis described in detail in the next section. It is interesting to note that significant amount of Ca<sub>2</sub>SiO<sub>4</sub> reacted in the initial phase of carbonation reaction leading to a reduction of 30.2% at breakthrough in as-is BOF slag. The Ca<sub>2</sub>SiO<sub>4</sub> was reduced by 37.3% and 80.5% in as-is BOF slag and fine BOF slag, respectively at the end of the experiment. The higher reactivity of Ca<sub>2</sub>SiO<sub>4</sub> in fine BOF slag

than as-is BOF slag which had relatively higher mean particle size than the fine slag further suggests surface area is a governing factor for the reactivity of the minerals. The significant increase in the calcite ( $\text{CaCO}_3$ ) content of the slag confirms the carbonation reactions in the slag. The percentage increase in  $\text{CaCO}_3$  content was greater than the percentage decrease in the free  $\text{CaO}$ ,  $\text{Ca(OH)}_2$  and  $\text{Ca}_2\text{SiO}_4$  together which means there were other calcium containing minerals contributing to the carbonation reactions. The total  $\text{CaCO}_3$  identified by QXRD in the as-is BOF slag at breakthrough, and as-is and fine BOF slag at the end of experiment correspond to 76.8, 111, and 278 g  $\text{CaCO}_3 \text{ kg}^{-1}$  BOF slag, respectively. Considering only free  $\text{CaO}$ ,  $\text{Ca(OH)}_2$  and  $\text{Ca}_2\text{SiO}_4$  reacted with  $\text{CO}_2$ , the reduction of their wt% as shown in **Table 4** corresponds to a removal of 47.6, 59.2 and 104.3 g  $\text{CO}_2 \text{ kg}^{-1}$  BOF slag and formation of 108.2, 134.5 and 237.0 g  $\text{CaCO}_3 \text{ kg}^{-1}$  BOF slag, respectively as per the reactions shown in **Eqs. 6, 7 and 8**.



The  $\text{CO}_2$  removal calculated stoichiometrically from the reduction in the free  $\text{CaO}$ ,  $\text{Ca(OH)}_2$  and  $\text{Ca}_2\text{SiO}_4$  in the slag is less than the experimentally observed values. The elemental composition of the slag showed significant portion of C in amorphous phase which means some of the carbonates could be in the amorphous phase. It further confirms the hypothesis that there are minerals other than the three above mentioned minerals responsible for carbonation reaction. The other calcium containing mineral which showed decreasing trend was Srebrodolskite ( $\text{Ca}_2\text{Fe}_2\text{O}_5$ ) as shown in **Table 4**. Similarly, the slag had significant amount of Ca in amorphous phase (nearly 35%, **Table 3**) which may also have been responsible for carbonation reactions.

Although the possible reactions of  $\text{H}_2\text{S}$  with steel slag suggest formation of sulfur,  $\text{FeS}$  and  $\text{Fe}_2\text{S}_3$  (**Eqs. 1, 2 and 4**), no  $\text{FeS}$  and  $\text{Fe}_2\text{S}_3$  were detected in the QXRD analysis which suggests they could be in the amorphous form (Rickard and Luther 2007). Around 1.15 wt% of crystalline sulfur was detected in the fine BOF slag which corresponds to  $\text{H}_2\text{S}$  removal of  $36.6 \text{ g H}_2\text{S kg}^{-1}$  BOF slag as per the stoichiometric relation shown in **Eq. 1** and is consistent with the  $\text{H}_2\text{S}$  removal capacity quantified from the column experiment (**Figure 6b**). It also confirms that the major reaction mechanism for  $\text{H}_2\text{S}$  removal is precipitation of sulfur. However, it is interesting that no crystalline sulfur was detected in the short term and long term exposed as-is BOF slag samples in the QXRD analysis whereas the elemental analysis shows the presence of sulfur both in crystalline and amorphous phase however, the crystalline phase is significantly lower (**Table 2**). The low crystalline sulfur content in those samples may have resulted in the low intensity peak which appear as a background peak in the search algorithm followed for QXRD thereby preventing the detection of the element.

### ***Morphological Changes in Slag Exposed to LFG***

**Figure 7** shows the composition based on SEM-EDS analysis of the virgin (fresh) as-is BOF slag, as-is BOF slag exposed to 20% (v/v) of  $\text{H}_2\text{S}$  in a batch reactor and coarse, as-is and fine BOF slag exposed to synthetic LFG in column reactor. The values represent average of 3-6 spectra (or locations) on a sample. The QXRD analysis did not show sulfur in coarse and as-is BOF slag exposed to synthetic LFG however, sulfur was detected in the EDS analysis. Although, absolute quantification cannot be done based on the EDS analysis results, it helps to give an idea on the compositional changes occurred on the sample surfaces. The BOF slag exposed to 20% (v/v)  $\text{H}_2\text{S}$

showed highest sulfur content which is reasonable as the slag surface was completely free for H<sub>2</sub>S absorption without competition for available sites for carbonation.

**Figures 8 (a-h)** show SEM images of virgin as-is BOF slag, coarse, as-is and fine BOF slag exposed to synthetic LFG in column reactor, and as-is BOF slag exposed to 20% (v/v) H<sub>2</sub>S in a batch reactor. All the SEM micrographs were taken at a scale of 1  $\mu$ m except for coarse BOF slag which showed significant charging effects at that scale hence the SEM image for coarse BOF slag corresponds to 10  $\mu$ m scale (**Figure 8b**). It is hard to differentiate between carbonate and sulfur crystals from the SEM micrographs as the samples were exposed to mixture of CO<sub>2</sub>, CH<sub>4</sub> and H<sub>2</sub>S in column reactors, however some distinct crystalline features were captured on the slag surfaces which were remarkably different from the carbonate crystal structures. The coarse and as-is BOF slag exposed to synthetic LFG showed rhombohedral crystals (**Figures 8b** and **8c**, respectively) covering the slag surface which can be identified as carbonate crystals based on previous studies (Reddy et al. 2019a, b, c). This further confirms the hypothesis that the formation of CaCO<sub>3</sub> on the slag surface could be limiting the interaction of CO<sub>2</sub> with the underlying minerals. Sulfur commonly exists in powdery form with fine acicular crystals which are very fragile and have the tendency to break and thus appear like powder (Sarbu et al. 2018). The fine BOF slag exposed to synthetic LFG (**Figure 8e**) and as-is BOF slag exposed to 20% (v/v) H<sub>2</sub>S (**Figure 8g**) showed powdery appearance covering the rhombohedral and plate like crystals which could be identified as sulfur crystals. Sarbu et al. (2018) reported twinning behavior in sulfur crystals obtained from the sulfur deposits in a sulfur cave. The as-is and fine BOF slag showed some twinning behavior (**Figure 8d** and **8f**, respectively) as observed by Sarbu et al. (2018). However, the twinned crystal structures in fine BOF slag were only observed at higher magnification (100 nm). Unlike slag exposed to synthetic LFG, the slag exposed to 20% (v/v) H<sub>2</sub>S showed two

different shapes of the crystals, fine acicular crystals (**Figure 8g**) which could be sulfur crystals (Xie et al. 2017, Sarbu et al. 2018) and plate like shape (**Figure 8h**) which could be iron sulfides (FeS and FeS<sub>2</sub>) (Xie et al. 2017; Shimizu et al. 2015). The fact that the slag with 20% (v/v) H<sub>2</sub>S was not exposed to any CO<sub>2</sub>, there is no possibility for the carbonate crystal formation on the slag surface which further suggests the plate like crystals on the slag surface to be of iron sulfide.

## Conclusions

The study explored the CO<sub>2</sub> and H<sub>2</sub>S removal potential of BOF slag under LFG conditions by carrying out series of batch and column studies. The study shows the BOF slag can remove CO<sub>2</sub> and H<sub>2</sub>S simultaneously when exposed to synthetic LFG at various moisture conditions. Effect of moisture is more prominent on H<sub>2</sub>S removal than on the CO<sub>2</sub> removal under similar moisture conditions. Once the slag is in a moist condition, increasing the moisture does not significantly increase the CO<sub>2</sub> removal potential whereas, H<sub>2</sub>S removal potential of the slag increases significantly with increasing moisture content showing the importance of water of hydration for H<sub>2</sub>S dissolution. The particle size of slag affects the CO<sub>2</sub> and H<sub>2</sub>S removal potential significantly as the slag showed increasing gas removal potential with decreasing particle size which was attributed to the high surface area and greater availability of the minerals for reaction. The mineralogical analysis of virgin BOF slag and BOF slags exposed to LFG under various conditions confirmed that apart from free CaO, Ca(OH)<sub>2</sub> and Ca<sub>2</sub>SiO<sub>4</sub>, there are other minerals which may participate in carbonation reactions. The major mechanisms for H<sub>2</sub>S removal involved precipitation of sulfur and formation of iron sulfides showing Fe as a responsible element for H<sub>2</sub>S removal. A maximum CO<sub>2</sub> removal of 300 g CO<sub>2</sub> kg<sup>-1</sup> BOF slag and H<sub>2</sub>S removal of 38 g H<sub>2</sub>S kg<sup>-1</sup>

<sup>1</sup> BOF slag was obtained for the BOF slag exposed to synthetic LFG. The assessment of behavior of the BOF slag under real field conditions is underway to further confirm the performance of the BOF slag for simultaneous removal of CO<sub>2</sub> and H<sub>2</sub>S under landfill conditions.

Using BOF slag in the landfill cover will not only mitigate the harmful emissions like CO<sub>2</sub> and H<sub>2</sub>S, but also provide meaningful use to the steel slags which are otherwise stockpiled or landfilled. Besides, the BOF slag can be used as a filter to purify the LFG for CH<sub>4</sub> recovery in the waste to energy plants. The BOF slag, if used in conjunction with biochar amended soil, can mitigate CH<sub>4</sub> along with CO<sub>2</sub> and H<sub>2</sub>S such as in the biogeochemical cover system recently introduced by Reddy et al. (2018). Hence, the future studies need to explore the optimization of the biogeochemical processes and develop a sustainable cover system for the landfills.

## **Acknowledgement**

This project is funded by the U.S. National Science Foundation (grant CMMI # 1724773), which is gratefully acknowledged. Any opinions, findings, and conclusions or recommendations expressed in this material are those of the authors and do not necessarily reflect the views of the National Science Foundation. Phoenix Services, LLC, is acknowledged for being an industrial partner on this project and providing slag samples for the experiments. Pittsburgh Mineral and Environmental Technology, Inc. and UIC Electron Microscopy Core facility are greatly acknowledged for providing services for sample analysis in this project.

## **Data Availability**

All data generated during the study appear in this article.

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**Table 1.** Summary of studies exploring use of steel slag to sequester H<sub>2</sub>S under various conditions

Material	Experimental Condition	Slag Particle Size	Max. H <sub>2</sub> S removal	Reference
Linz-Donawitz Steel Slag	Reactor: fixed bed quartz reactor Gas condition: 1000 ppmv (1.39 mg H <sub>2</sub> S/L) H <sub>2</sub> S in moist air (RH: 10 and 50%) Temperature: Room temperature (25 ± 2 °C)	< 212 mm; 212-500 mm	180 g H <sub>2</sub> S/kg slag	Montes-Morán et al. (2012)
Steel making slag (slag type not mentioned)	Reactor: 150 mL vial bottle, agitated at 60 rpm Gas condition: H <sub>2</sub> S solution with a concentration ranging from 0 to 200 mg S/L Temperature: 25 °C	2-5 mm	37.5 g S/kg slag	Kim et al. (2012)
Steel making slag-carbonated) (slag type not mentioned)	Reactor: 100 mL vial bottle, agitated at 100 rpm Gas condition: H <sub>2</sub> S solution with a concentration of 10 to 100 mg/L Temperature: 25 °C	0.8-5 mm	7.5 g H <sub>2</sub> S/kg slag	Asaoka et al. (2013)
BOF slag	Reactor: Batch reactor, agitated at 300 rpm Gas condition: 20 % (v/v) H <sub>2</sub> S Moisture condition: 0, 20% and 40% (w/w) Temperature: Room temperature (20 °C)	<1 mm, 1-6 mm	119 g H <sub>2</sub> S/ kg slag	Sarperi et al. (2014)
BOF Slag	Reactor: Anaerobic digester, slag added at dosages of 1 g/L to 50 g/L Gas condition: Digester biogas Temperature: 32 ± 2 °C	<1.6 mm	78% H <sub>2</sub> S removal	Caicedo-Ramrez et al. (2020)
BOF slag	Reactor: Batch reactor (manual stirring) and fixed column reactor, Gas condition: Synthetic LFG (50% CO <sub>2</sub> , 48.25% CH <sub>4</sub> and 1.75% H <sub>2</sub> S); 20 % H <sub>2</sub> S Moisture condition: 0, 10%, 20% and 30% (w/w) Temperature: Room temperature (25 °C)	<0.075 mm, <0.075-4.75 mm, 0.425-4.75 mm	38 g H <sub>2</sub> S/kg slag	Current study

**Table 2.** Normalized elemental composition (wt%) of as-is and fine BOF slags

Element	Fresh As-is		Breakthrough As-is (PV = 100)		Long term exposed As-is (PV = 1,900)		Long term exposed Fines (PV 4,950)	
Ca	43.3	(A 37.3%) (C 62.7%)	42.2	(A 41.8%) (C 58.2%)	41	(A 36.7%) (C 63.3%)	41.8	(A 26.5%) (C 73.4%)
C	0.8	(A 71.8%) (C 28.2 %)	2.6	(A 39.1%) (C 60.9 %)	4	(A 34.9%) (C 65.1%)	8.55	(A 10.7%) (C 89.3%)
S	0.1	(A 77.2%) (C 22.8%)	0.4	(A 80.2%) (C 19.8%)	1.1	(A 91.5%) (C 8.5%)	2.9	(A 0%) (C 100%)
Fe		32.7		32.1		31.9		25.15
Mg		10.2		10.1		9.7		10
Si		7		7.2		7.5		8.6
Mn		2.6		2.6		2.6		2.45
Al		1.2		1.2		1.2		1.1
Ti		0.4		0.4		0.4		0.4
P		0.3		0.3		0.3		0.3
Cl		0.1		0.1		0.1		0.2
V		0.1		0.1		0.1		0.1
Na		0.1		0.1		0.1		0
H		0.3		0		0		0
F		0		0		0		0
Zr		0		0		0		0

**Note:** PV = pore volume; A = amorphous fraction; C = crystalline fraction

**Table 3.** Major oxide chemistry of as-is and fine BOF slags

<b>Major oxides</b>	<b>Fresh</b>	<b>Breakthrough As-is (PV = 100)</b>	<b>Long-term As-is (PV = 1,900)</b>	<b>Long-term Fines (PV 4,950)</b>
CaO	38.5	36.6	34.9	31.3
SiO <sub>2</sub>	10.4	9.9	9.4	7.95
Al <sub>2</sub> O <sub>3</sub>	1.5	1.4	1.4	1.1
Fe <sub>2</sub> O <sub>3</sub>	29.7	28.4	27.7	19.25
MgO	10.7	10.4	9.8	8.9
SO <sub>3</sub>	0.2	0.6	1.7	-
LOI	4.8	8.8	11.4	24.5

**Note:** PV = pore volume



**Table 4.** Mineralogical evolution of the as-is and fine BOF slags

Mineral	Formula	Fresh As-is	Breakthrough As-is (PV = 100)	Long-term As-is (PV = 1,900)	Long-term Fines (PV 4,950)
Amorphous		30.7	30.3	33	40.6
Srebrodolskite	$\text{Ca}_2\text{Fe}_2\text{O}_5$	17.41	15.62	14.3	7.3
Larnite	$\text{Ca}_2\text{SiO}_4$	16.44	11.47	10.3	3.2
Iron Magnesium Oxide	(MgO)0.593 (FeO)0.407	9.58	16.09	9.3	6.5
Wuestite	$\text{FeO}$	9.19	6.66	8.4	3.15
Magnesioferrite	$\text{MgFe}_2\text{O}_4$	7.79	7.57	8.6	6.5
Lime	$\text{CaO}$	1.76	1.06	1	0
Portlandite	$\text{Ca(OH)}_2$	4.67	1.86	1	0.85
Calcite	$\text{CaCO}_3$	0.43	6.19	9.6	27.8
Vaterite	$\text{CaCO}_3$	0.75	1.49	1.5	-
Dolomite	$\text{CaMg(CO}_3)_2$	-	-	-	1.7
Sulfur	$\text{S}$	-	-	-	1.15

**Note:** PV = pore volume

## Figure Captions

**Figure 1.** Particle size distribution of the virgin BOF slags: coarse (#4-#60); as-is (original gradation); and fine (passing #140).

**Figure 2.** Simultaneous removal of synthetic LFG components by the as-is BOF slag at 0, 10, 20 and 30% (w/w) moisture contents: (a) CO<sub>2</sub>; (b) CH<sub>4</sub> (b); and (c and d) H<sub>2</sub>S. Solid lines denote gas concentration and dotted lines represent gas removal. Gas concentration 1% (v/v) = 10,000 ppmv

**Figure 3.** Removal of H<sub>2</sub>S by BOF slag in long term at different moisture contents: (a) variation of concentration and percentage removal with time; (b) cumulative removal at different points of time. Gas concentration 1% (v/v) = 10,000 ppmv

**Figure 4.** CO<sub>2</sub> removal by BOF slag in short-term (until breakthrough) and in long-term both at 10% (w/w) initial moisture content and continuous flow of synthetic LFG (48.25% CH<sub>4</sub>, 50% CO<sub>2</sub> and 1.75% H<sub>2</sub>S).

**Figure 5.** (a) CO<sub>2</sub> and (b) H<sub>2</sub>S gas removal in terms of ratio of outlet and inlet gas concentrations as a function of particle size of BOF slag for an initial moisture content of 10% (w/w) and continuous flow of synthetic LFG (48.25% CH<sub>4</sub>, 50% CO<sub>2</sub> and 1.75% H<sub>2</sub>S).

**Figure 6.** (a) Cumulative CO<sub>2</sub> and (b) H<sub>2</sub>S gas removal as a function of particle size of BOF slag for an initial moisture content of 10% (w/w) and continuous flow of synthetic LFG (48.25% CH<sub>4</sub>, 50% CO<sub>2</sub> and 1.75% H<sub>2</sub>S).

**Figure 7.** SEM-EDS based elemental composition of the various BOF slags including exposed to 20% (v/v) H<sub>2</sub>S in a batch reactor and synthetic LFG (48.25% CH<sub>4</sub>, 50% CO<sub>2</sub>, and 1.75% H<sub>2</sub>S) in column reactors based on SEM-EDS analysis.

**Figure 8.** SEM images of (a) virgin BOF slag; (b) coarse BOF slag exposed to synthetic LFG in a column reactor; (c and d) as-is BOF slag exposed to synthetic LFG in a column reactor; (e and f) fine BOF slag exposed to synthetic LFG in a column reactor; (g and h) as-is BOF slag exposed to 20% (v/v) H<sub>2</sub>S in a batch reactor.

