Biodegradation of Functionalized Nanocellulose Benjamin P. Frank<sup>1</sup>, Casey Smith<sup>1</sup>, Emily R. Caudill<sup>2</sup>, Ronald S. Lankone<sup>1</sup>, Katrina Carlin<sup>1</sup>, Sarah Benware, <sup>2</sup> Joel A. Pedersen<sup>2,3</sup>, D. Howard Fairbrother<sup>1\*</sup> <sup>1</sup>Department of Chemistry, Johns Hopkins University, 3400 N Charles Street, Baltimore, MD 21218, United States <sup>2</sup>Department of Chemistry, University of Wisconsin-Madison, 1101 University Avenue, Madison, Wisconsin 53706, United States <sup>3</sup>Departments of Soil Science and Civil & Environmental Engineering, University of Wisconsin-Madison, 1525 Observatory Drive, Madison, WI, 53706, USA \*Corresponding Author Email: howardf@jhu.edu Address: Department of Chemistry, 3400 North Charles Street, Baltimore, MD 21218 Keywords: Anaerobic digestion, surface chemistry, nanoparticle, biomethane potential tests, esterification, modified Gompertz model, degree of substitution 

**Abstract:** Nanocellulose has attracted widespread interest for applications in materials science and biomedical engineering due to its natural abundance, desirable physicochemical properties, and ease of mineralization (i.e., complete biodegradation). A common strategy to increase dispersibility in polymer matrices is to modify the hydroxyl groups on nanocellulose through covalent functionalization, but such modification strategies may affect the desirable biodegradation properties exhibited by pristine nanocellulose. In this work, cellulose nanofibrils (CNFs) functionalized with a range of esters, carboxylic acids, or ethers exhibited decreased rates and extents of mineralization by anaerobic and aerobic microbial communities compared to unmodified CNFs, with etherified CNFs exhibiting the highest level of recalcitrance. The decreased biodegradability of functionalized CNFs depended primarily on the degree of substitution at the surface of the material rather than within the bulk. This dependence on surface chemistry was attributed not only to the large surface area-to-volume ratio of nanocellulose, but also to the prerequisite surface interaction by microorganisms necessary to achieve biodegradation. Results from this study highlight the need to quantify the type and coverage of surface substituents in order to anticipate their effects on the environmental persistence of functionalized nanocellulose.

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**Synopsis:** This study demonstrated that the microbial biodegradation and environmental persistence of functionalized nanocellulose will be strongly influenced by the type and degree of surface functionalization with bulk functionalization playing a secondary role.

#### Introduction

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occurring biopolymer Nanocellulose, naturally consisting β-1.4-Da anhydroglucopyranose monomer units, is derived from macrocellulose via chemical treatment, 2 sonication,<sup>3</sup> mechanical milling,<sup>4</sup> or enzymatic digestion.<sup>5</sup> Nanocellulose possesses desirable mechanical properties (e.g., Young's modulus, tensile strength) comparable to Kevlar and steel.<sup>6</sup>, <sup>7</sup> These mechanical properties, along with the nanoscale width, natural abundance, biodegradability, and biocompatibility of nanocellulose, elevate its use in a variety of applications, including polymer nanocomposites, <sup>1, 2, 8</sup> biomedicine, <sup>9</sup> sensors, <sup>9, 10</sup> water treatment, <sup>9, 11</sup> coatings, <sup>12</sup> and smart materials. 13-15 Many applications of nanocellulose require hydrophobic surface modification (e.g., coating, functionalization) to improve its dispersion in organic media and reduce hydrogen bond-induced homoaggregation prior to use in material applications. 10, 16-18 Roughly 35 million tons of nanocellulose are produced globally each year, and production is projected to further increase by 2030.<sup>19, 20</sup> Release of nanocellulose composite materials into the environment is therefore inevitable, necessitating understanding the effect of surface modification on its microbial mineralization.

Products featuring cellulosic materials often advertise the complete biodegradability (i.e., mineralization of carbon into CO<sub>2</sub> and/or CH<sub>4</sub>) of cellulose as an attractive feature compared to traditional carbon-based reinforcement options such as carbon nanotubes and carbon fibers due to reduced environmental persistence and impact. The biodegradation of cellulosic materials proceeds through different mechanisms and microorganisms in anaerobic and aerobic environments. Specifically, in aerobic environments, cellulose is generally degraded by cellulase and  $\beta$ -glucosidase enzymes secreted by bacteria and fungi. Cellulases initiate degradation of the cellulose structure by hydrolyzing internal bonds (endoglucanases) and chain-ends

(cellobiohydrolases) to yield cellobiose molecules.<sup>25</sup> β-Glucosidase then concludes the depolymerization process by converting cellobiose into glucose which is mineralized to CO<sub>2</sub> and water by aerobic microorganisms.<sup>25</sup> In contrast, anaerobic microorganisms utilize cellulosomes or multiprotein complexes of enzymes to mineralize cellulose into water and biogas consisting of CO<sub>2</sub> and CH<sub>4</sub>.<sup>25-27</sup> This conversion to biogas is achieved at over 80% efficiency for cellulose, demonstrating the high biodegradability of the material.<sup>28</sup> Furthermore, while a single aerobic microorganism species is sufficient to fully mineralize cellulose (e.g., the fungus *Trichoderma reesei*), multiple anaerobe species are required to work in concert to produce the enzymes necessary for conversion of cellulose to biogas.<sup>26</sup> Examples of bacterial phyla responsible for cellulose degradation include Acidobacteria and Firmicutes.<sup>29,30</sup>

While native cellulose is readily fully biodegraded (mineralized), hydrophobic modifications have the potential to interfere with the enzymatic degradation of macrocellulose, <sup>31-33</sup> a process which depends on the composition and activity of the microbial community involved (e.g., aerobic vs. anaerobic). <sup>23, 25, 26, 29, 30, 34</sup> In a previous report, we demonstrated that this interference held true for cellulose nanofibrils (CNFs) modified with hydrophobic siloxane coatings, which blocked enzymatic access to nanocellulose and inhibited its anaerobic mineralization. <sup>16</sup> In contrast, covalent functionalization strategies utilizing ether, ester, and urethane linkages avoid the formation of surface coatings, and have been widely applied to macrocellulose and nanocellulose to improve dispersion in organic media and polymers, <sup>35-38</sup> yet their impact on nanocellulose biodegradation has not been previously investigated. Results from previous studies lead to the expectation that for functionalized nanocellulose, the rate limiting step in biodegradation will involve the removal of functional groups (e.g., ester groups by hydrolysis) to regenerate the functionalizing reagent and the hydroxyl groups present in native cellulose. <sup>32, 39-1</sup>

<sup>41</sup> After this initial cleavage, the biodegradation pathway of functionalized cellulose proceeds through biodegradation of the native cellulosic component and the functionalizing reagent.

The most commonly used metric to express the extent of covalent functionalization of cellulose is the degree of substitution (DS), representing the average number of cellulosic hydroxyl groups functionalized per anhydroglucose monomer unit (DS = 0-3). The conventionally determined DS value represents the extent of functionalization of both the surface and bulk regions of the material and can therefore be regarded as a measure of the overall DS (DSoverall). For cellulosic materials, the DSoverall is generally determined using elemental analysis<sup>42</sup> or nuclear magnetic resonance (NMR) spectroscopy.<sup>43, 44</sup> For covalently modified macrocellulose, biodegradability depends on both DSoverall and the nature of the chemical linkage (i.e., ether, <sup>45-47</sup> ester<sup>31, 32, 48</sup>). For example, degradation of macrocellulose fibers functionalized with carboxymethyl groups (ether linkage) by a cellulolytic enzyme complex decreased as the DSoverall increased from 0.41 to 1.30.<sup>46</sup> Furthermore, nanocellulose esterified with acetyl groups to DSoverall >1.25 exhibited significant inhibition of anaerobic biodegradation as compared to un-modified macrocellulose.<sup>48</sup>

Past studies typically quantified the extent of modified macrocellulose biodegradation in terms of the production of low molecular mass byproducts (e.g., cellobiose, glycolic acid) rather than evolution of CO<sub>2</sub> or CH<sub>4</sub>. Furthermore, many studies on modified macrocellulose employed model enzymes (e.g., cellulase, esterase) or a single microbial species to effect biodegradation.<sup>46, 47, 49</sup> While the information from these studies is useful in identifying trends in biodegradation as a function of material properties, such approaches do not measure complete biodegradation of the cellulosic material and do not represent the microbial communities encountered in natural environments.<sup>23, 24, 34, 46, 47, 49</sup> Failure to discern the complete mineralization of functionalized

cellulose has led to disagreement with respect to the degree of inhibition resulting from chemical functionalization.<sup>31,33,45</sup> Additionally, as mineralization of cellulosic materials generally proceeds via the cooperation of a microbial community,<sup>27</sup> more complex systems utilizing environmentally relevant microorganisms are best suited for assessing the environmental persistence of functionalized nanocellulose, rather than model enzymes or pure microbial cultures.<sup>23,34</sup> One established method of measuring the anaerobic biodegradation of cellulose involves the quantification of biogas produced during the mineralization of carbon into CO<sub>2</sub> and CH<sub>4</sub>.<sup>28,31,48,50</sup> The aerobic biodegradation of cellulose is typically quantified using mass loss measurements to compare the amount of carbon converted from the solid (i.e., cellulosic) phase into the gas phase (i.e., CO<sub>2</sub>).<sup>31,51,52</sup>

Another potentially important factor to consider is that the extent to which the biodegradation of a functionalized nanomaterial is inhibited may be more closely linked to the degree of functionalization of the surface (DS<sub>surface</sub>) than to DS<sub>overall</sub>. This distinction is important as the preliminary step in the biodegradation of a solid-phase material involves the adsorption and colonization of microorganisms at the surface.<sup>53-56</sup> In the case of cellulosic materials, this initial biodegradation step requires biofilm formation or the interaction of highly specific microbesecreted cellulosome complexes with its surface.<sup>26, 57-59</sup> As nanocellulose fibers are composed of numerous cellulose chains woven together into a nano-scale cord, the chains at the fiber surface are distinct from those within the bulk of the material.<sup>2, 60, 61</sup> During chemical functionalization with liquid reagents, cellulose chains in both the bulk and surface of nanocellulose are targeted,<sup>2, 62, 63</sup> while gas phase reagents selectively functionalize the nanocellulose surface due to their inability to penetrate into the bulk of the material.<sup>64-66</sup> Despite the potential for achieving different levels of surface vs. bulk functionalization, studies of cellulosic materials typically use only bulk-

sensitive analytical techniques (e.g., NMR), and thus quantify only DS<sub>overall</sub>.<sup>42, 44, 67-70</sup> The effect of surface substitution is likely to be particularly important for the biodegradation of CNFs compared to macrocellulose due to the large surface area-to-volume ratio of nanocellulose as well as the decreased swelling capacity of CNFs which limits access to bulk cellulose chains.<sup>2</sup>

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In this study, we compare the influence of surface vs. bulk functionalization as well as the influence of different covalent linkages on CNF mineralization by aerobic and anaerobic microbial communities. This study is the first to investigate the biodegradability of a range of functionalized nanocellulose in both aerobic and anaerobic environments. Selective functionalization of the surface and bulk regions was accomplished using liquid-phase and gas-phase (i.e., surfacespecific)<sup>64,65</sup> techniques to esterify nanocellulose with long-chain hydrocarbons that are often used to improve CNF dispersion in polymer nanocomposites.<sup>63, 71</sup> Attenuated total internal reflectance Fourier-transform infrared spectroscopy (ATR-FTIR), solid-state <sup>13</sup>C-nuclear magnetic resonance spectroscopy (<sup>13</sup>C-NMR), and CHN elemental analysis were used to confirm functionalization. Elemental analysis was used to determine DS<sub>overall</sub>, while X-ray photoelectron spectroscopy (XPS) was utilized to measure DS<sub>surface</sub>. <sup>72</sup> To assess the effect of different covalent linkages, CNFs were functionalized with different esters (phenyl, hexyl, dodecyl) and ethers (hexyl, dodecyl), which were also compared to common TEMPO oxidized nanocellulose carboxylates with H<sup>+</sup> or Na<sup>+</sup> counterions. Biodegradation of these samples by anaerobic and aerobic microorganisms was assessed via biomethane potential (BMP) tests and mass loss, respectively. Results from our study reveal the importance of materials characterization, particularly the surface coverage of added functional groups, in understanding the biodegradation behavior of functionalized CNFs.

## **Experimental**

Functionalization of CNFs. Freeze-dried cellulose nanofibrils (CNFs) were purchased from the University of Maine Process Development Center and either used as-received or milled into a powder with a Flack-tek mill (DAC 150, 2800 rpm, 4 min). Ethyl cellulose (48.0/48.5 % w/w ethoxyl basis) was purchased from Sigma-Aldrich. Carboxylated CNFs were purchased from the University of Maine Process Development Center as a slurry of TEMPO-oxidized CNFs (1 wt% CNF, 1.5 mmol COOH/g cellulose) and dried either as-received (Na<sup>+</sup> counterion) or after washing with dilute HCl (H<sup>+</sup> counterion).

Esterified CNFs were prepared by liquid-phase reactions with carboxylic acid reagents, <sup>73</sup> or with acyl chloride reagents. <sup>63</sup> CNF functionalization with carboxylic acids was performed by dispersing 200 mg of CNFs in 200 mL deionized water followed by a 2 h sonication before adjusting to approximately pH 4 with 4 M HCl. The mixture was then heated to evaporate water followed by addition of excess phenyl acetic acid (phenyl ester CNF), hexanoic acid (hexyl ester CNF), or dodecanoic acid (dodecyl ester CNF, DA-CNF) before melting at ~140 °C to form the reaction medium. Sample solutions were stirred with a magnetic stirrer for 14 h and subsequently quenched with ethanol (Pharmco, 200 proof ACS/USP grade). Functionalized CNF powders were recovered via vacuum filtration, washed with ethanol, and dried in a vacuum oven at 60 °C for 12 h.

Liquid-phase esterification reactions using acyl chlorides were carried out using a modified method derived from literature by dispersing 200 mg of CNFs in 12 mL of diethyl ether and 0.5 mL of triethylamine in a vented round bottom flask equipped with a magnetic stirrer.<sup>63</sup> After dropwise addition of 1 mL of lauroyl chloride, samples were gently mixed at room temperature for 6 h. At the end of the reaction time, samples were quenched with 30 mL of deionized water and recovered

by vacuum filtration followed by a dilute HCl (100 mL, pH 5.5) and a deionized water (800 mL) wash. Samples were then dried in a vacuum oven at 50 °C for 72 h to yield lauroyl chloride esterified CNFs (LC-CNF).

Gas-phase (GP) esterification was performed by adding ~10 mg of CNF powder to a custom-designed Schlenk line vessel<sup>74</sup> suspended above 1 mL of either lauroyl chloride (GP-LC-CNF) or hexanoyl chloride (GP-HC-CNF). The bottom of the vessel was submerged in liquid nitrogen to freeze the reagent, followed by headspace evacuation. After sealing the vessel, the reagent was allowed to thaw and vaporize into the headspace of the vessel to react with the CNF powder.

Etherification was performed by swelling 200 mg of dried CNF in 200 mL of dimethyl sulfoxide (DMSO, Fisher, 99.9%) via sonication for 3 h. After swelling, 200 mg K<sub>2</sub>CO<sub>3</sub> (Aldrich, 99.99%) was then added, and the sample sonicated for an additional 3 h. A 30 mL aliquot of 1-bromohexane (hexyl ether CNF; Aldrich, 98%) or 1-bromododecane (dodecyl ether CNF; Aldrich, 97%) was added to the sample before heating to 90 °C and magnetically stirring for 45 min under reflux.<sup>75</sup> The reaction was then quenched with ethanol and the functionalized CNF powder recovered via vacuum filtration followed by thorough washing with ~1 L of ethanol (Pharmco, 200 proof ACS/USP grade) before being dried in a vacuum oven at 60 °C for 12 h.

**CNF** Characterization. Cellulose nanofibril characterization techniques are briefly described, with complete details in the SI. Functional groups in the unmodified and functionalized CNFs were identified using ATR-FTIR; the bonding and concentrations of C and O at the surface of the unmodified and functionalized CNFs were assessed using XPS; the carbon structure of the nanocellulose before and after functionalization was evaluated via solid-state <sup>13</sup>C-NMR; the wt% C, N, O, and H of unmodified and functionalized CNFs was determined by elemental analysis.

**Degree of Substitution (DS) Calculations.** *DS from Elemental Analysis*. For CNFs functionalized with esters and ethers DS values were calculated from the wt % carbon (Table S1) relative to unmodified CNF (C<sub>6</sub>H<sub>10</sub>O<sub>5</sub>)<sup>44, 73, 76</sup> with an uncertainty of approximately 0.3 %.<sup>77</sup> For example, an increase in carbon content to 53.0 wt % after esterification with dodecanoic acid (C<sub>12</sub>H<sub>24</sub>O<sub>2</sub>) reflects a DS of 0.45, which corresponds to an average addition of approximately one dodecyl ester group per two glucose monomer units. Because elemental analysis measures the degree of CNF functionalization from the entirety of the sample, DS values determined from elemental analysis are hereafter referred to as DS<sub>overall</sub>.

DSoverall of TEMPO CNF. The TEMPO CNF obtained from University of Maine Process Development Center was listed as having 1.5 mmol COOH/g cellulose. Each gram of cellulose features roughly 6.2 mmol of glucose monomer units, which corresponds to 0.243 COOH groups per cellulose unit (1.5 mmol COOH/6.2 mmol cellulose), representing a DSoverall of 0.24.

DS from XPS. Degree of substitution values determined by XPS for CNF esters and ethers were based on the fitted contribution from the C–C component (285.0 eV) to the C(1s) XPS envelope. As the C–C content in unmodified CNF was  $14.5 \pm 3\%$  (due to adventitious carbon), any increase was assumed to be due to functionalization of the nanocellulose surface by ethers or esters. For example, upon esterification with dodecyl ester groups, an increase in the C–C component to 48% would require an average addition of approximately 1 dodecyl ester group per 7 glucose monomer units, corresponding to a DS of 0.43. Since DS values determined by XPS are surface specific and represent the degree of CNF functionalization within only the topmost 2 nm to 3 nm of the sample, they are hereafter referred to as DS<sub>surface</sub> (Table S2).

DS for Gas-Phase CNF Samples. We estimated the degree of substitution for gas-phase CNF samples (GP-HC-CNF and GP-LC-CNF) from a combination of CHN analysis data and

ATR-FTIR spectra. Due to the low sample mass attainable from gas phase functionalization, we were not able to measure CHN on the full set of samples. Instead, the elemental composition of one sample from each set (GP-HC-CNF-4 and GP-LC-CNF-4) was determined, and the DSoverall calculated as described above. The DSoverall values of GP-HC-CNF-4 and GP-LC-CNF-4 were then ratioed to the C=O (ester): C-O (cellulose) peak intensities obtained from ATR-FTIR analysis. As detailed in the results and discussion section, this provided a means to convert the C=O: C-O peak intensities measured on the remaining samples in the two series (GP-HC-CNF and GP-LC-CNF) into their respective DSoverall values.

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Anaerobic Biodegradation of CNFs. Biomethane Potential (BMP) Tests. Mineralization was assessed using biomethane potential tests, adapted with minor modifications from Owen et al., 50, 78 to monitor biogas (CO<sub>2</sub> and CH<sub>4</sub>) production after unmodified and functionalized CNFs were incubated with an anaerobic microbial community. Microbial BMP media (Table S3) was prepared as previously described<sup>16</sup> and heated at 100 °C for 30 min while sparging with N<sub>2</sub> to achieve anoxic conditions before adding anaerobic digestor sludge (10% v/v) obtained from Back River Wastewater Treatment Plant (Baltimore, MD). The BMP media was adjusted to pH ~7 using 20% CO<sub>2</sub> gas and kept anoxic via continuous N<sub>2</sub> sparging. Duplicate 100 mg or 150 mg (DA-CNF only, due to increased sample availability) functionalized CNF samples were mixed with 100 mL of anaerobic media in 150 mL serum bottles and capped with rubber septa. We also assessed 150 mg of each functionalizing reagent (e.g., dodecanoic acid, hexanoic acid) independently to determine their biogas production potential. Samples were incubated at mesophilic temperature (35 °C) for up to 424 d, and biogas production was volumetrically assessed via intermittent headspace measurements with a glass syringe. In each set of samples, blank controls (i.e., anaerobic media including the same concentration of sludge in the absence of a CNF sample) were

incubated in triplicate to account for biogas produced by the residual organic matter in the media (< 10% total solids, ~55% volatile solids before dilution to 10% of media volume).<sup>79-81</sup> Separate control studies were performed with the native (i.e., unfunctionalized) CNF to determine the extent of biogas production in the absence of functionalization and to confirm that the overwhelming majority of CNFs biodegrade to liberate biogas. Importantly, the carbon contributed by cellulose in each sample (42 mg C per 100 mg unfunctionalized CNF, more for functionalized CNF) vastly outweighed the contributed carbon from the BMP media (< 5 mg C in nutrients, most of which is not mineralized). Given the well-known propensity of cellulose to form biogas during biodegradation, the biogas produced by CNF samples was dominated by CNF mineralization. To account for biogas produced from the BMP media, the biogas production from CNF samples at each timepoint was reported as the difference between the volume produced by the CNF sample (typically yielded > 5 mL at each time point) and the average volume of biogas produced from the blank media (< 3 mL per timepoint). In this way, any biogas contribution from the media is removed and the reported biogas data arises solely from the mineralization of the CNF sample. All biogas values were normalized to account for differences in sample mass (comparison between biogas production from 100 mg and 150 mg CNF; Figure S1).

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Biodegradation of CNFs in our BMP tests lead predominantly to biogas formation over the course of a few weeks, producing between 680-700 mL/g of biogas, representing over 80% CNF mass loss. This is consistent with the rapid and extensive mineralization of cellulosic materials typically observed.<sup>28, 82</sup> Despite the efficiency of biogas production during CNF biodegradation, some of the carbon is converted into biomass, as is typical of biodegradation processes. This situation also holds true for functionalized CNFs.

The biodegradation of functionalized CNFs proceeds via the hydrolytic cleavage of the

linkage formed between the functional group and the cellulosic monomer, and the subsequent biodegradation of the species generated in this step. For example, in the case of the CNFs functionalized with acyl chlorides, the process will be initiated by the generation of the native CNFs and carboxylic acids, followed by their subsequent biodegradation. For each functionalized CNF, we therefore performed independent biodegradation studies to determine the partitioning between biogas and biomass production for each component, after subtraction of the small amount of biogas produced due to residual carbon and biomass present in the media itself (determined in separate control studies). This information enabled us to determine the biogas each component would generate in the case of complete biodegradation under our experimental conditions. Combined with knowledge of the chemical composition of each functionalized CNF, we could then determine the biogas we would predict in the event of complete biodegradation. For example, a CNF functionalized with a dodecyl ester with a DS<sub>overall</sub> of 0.45 would be composed of roughly 66 wt % CNF and 34 wt % dodecyl ester. The total biogas produced from this functionalized CNF in the event of complete (100% biodegradation) is expected to be 0.66x + 0.34y, where x and y are the per gram biogas production potentials of cellulose (680 mL g<sup>-1</sup>) and dodecanoic acid (1280 mL g<sup>-1</sup>), respectively (Figure S2). This equates production of 883 mL/g biogas, considerably more than produced from cellulose alone (i.e., 680 mL/g). This predicted value was nearly met for DA-CNF-2 (94% of calculated biogas production was achieved), providing evidence for almost complete biodegradation of this sample, thereby also providing support for the validity of this normalization strategy in computing biogas potentials. By reporting data in these normalized terms (i.e., experimental data/calculated maximum data), we were able to compare samples in terms of their ability to achieve "maximum" biodegradation (a normalized  $V_{\text{max}}$  value of 1). Explicitly, this value represents the extent to which each functionalized CNF reaches its maximum biogas production

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based on the overall partitioning predicted for biogas and biomass production from the components. The biogas values that would be produced in the event of 100% mineralization along with the experimentally measured (empirical) biogas produced for CNFs and all of the functionalized CNFs used in this study are reported in Table S4.

Gompertz Modeling. Anaerobic biogas production rates were quantified using the modified Gompertz kinetic model (Eq. 1)<sup>83-85</sup>

$$V_i = V_{max} e^{-e^{(\frac{K*e(\lambda - t_i)}{V_{max}} + 1)}}$$
 (Eq. 1)

where  $V_{\text{max}}$  is the experimental ultimate biogas yield (mL g<sup>-1</sup>), K is the maximum specific rate constant (mL g<sup>-1</sup> d<sup>-1</sup>),  $\lambda$  is the lag phase time constant (d), and  $t_i$  is the total incubation time (d). The Solver optimization tool in Microsoft Excel was used to estimate the model parameters for each sample by minimizing the root mean square deviations (RMSE, Table S5), and the agreement between predicted and experimental values was evaluated by comparing the RMSE and  $R^2$  values.

Aerobic Biodegradation Tests. Aerobic biodegradation of CNF samples was assessed by measuring mass loss after exposure to an aerobic microbial community obtained from the primary effluent of the Back River Wastewater Treatment Plant (Baltimore, MD). Mass loss was used as the metric for aerobic biodegradation as an open system was required to maintain an oxygenated environment. Triplicate 50 mg samples of CNF powders were sedimented via centrifugation in conical vials (Figure S3) containing an aqueous medium composed of 200 mg/L sodium acetate trihydrate and 10% v/v salt stock (7.18 mM K2HPO4, 2.79 mM KH2PO4, 0.757 mM (NH4)2SO4, 0.0406 mM MgSO4•7H2O), and trace elements necessary for bacterial growth. Microbial media was made by adding 10% v/v primary effluent supernatant to the vials and shaking at 125 rpm at 28 °C for 60 d. These samples were then incubated for 60 days before the powders were recovered

from the media, washed with ultrapure water (18.2 M $\Omega$ ·cm, Millipore, USA), washed three times with ethanol to remove any adhered biomass/biofilm and then dried in a vacuum oven at 60 °C for 12 h to evaporate any adsorbed water, before being weighed. This approach is analogous to the one used in other aerobic biodegradation studies of cellulosic materials. <sup>28, 51, 52</sup> To account for any native material which was dispersed or otherwise lost in the media, an identical set of samples was incubated for 60 days in uninoculated media and the mass loss observed in these control studies was subtracted from the biotic mass loss values. Consistent with our expectations, the mass loss experienced by the native CNF samples was reproducible and close (80%) to the mass of the CNF added, supporting the idea that comparisons between the extent of mass loss produced by different samples could serve as the basis to compare the extent of biodegradation amongst our functionalized CNF samples. Furthermore, the products of incomplete CNF biodegradation (i.e., cellulose hydrolysis without complete conversion to CO<sub>2</sub>) such as cellobiose and glucose monomer units are water soluble and would therefore contribute to the observed mass loss. 86 Thus, the final mass measured in our studies should be composed predominately of undegraded CNF or functionalized CNF samples, as intended. The mass loss for each sample was determined by the difference between the average mass lost in bacterial culture minus the average mass lost in the abiotic media. This difference was then ratioed to the initial mass (50mg) to determine the adjusted % mass loss reported in Figure 1.

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In the present investigation, mass loss data from aerobic biodegradation studies was used as a semi-quantitative measure of biodegradation amongst functionalized CNFs. Unlike the BMP tests, mass loss was determined at a single time point, precluding application of the modified Gompertz model. Moreover, in the aerobic studies we did not directly assess the mass loss that would accompany the biodegradation of the added functional groups, although the influence of

this unknown is not expected to impact the qualitative conclusions drawn from these studies.

#### **Results and Discussion**

CNF naming convention: CNFs were functionalized with different ester, carboxylic acid, and ether groups (Figure S4, S5). CNF ethers functionalized using 1-bromohexane and 1-bromododecane are referred to as hexyl ether CNF and dodecyl ether CNF, respectively. CNFs functionalized using phenyl acetic acid, hexanoic acid, and dodecanoic acid (DA) are referred to as phenyl, hexyl, and dodecyl ester CNF, respectively. Samples functionalized with liquid-phase DA are referred to as DA-CNF-X, where X represents the relative DS<sub>surface</sub> rank in that sample series. For example, DA-CNF-4 represents a CNF functionalized with liquid phase DA at the highest DS<sub>surface</sub> value of the four samples within the series. Similarly, CNFs functionalized with liquid phase lauroyl chloride (LC) are referred to as LC-CNF-X, and follow the same naming convention. Functionalizations achieved using gas-phase (GP) acyl chlorides are denoted by a GP naming scheme. For example, the CNF with the surface most extensively functionalized by gas phase hexanoyl chloride (HC) in a series of four is labeled GP-HC-CNF-4.

Figure 1a shows that unmodified CNFs were completely mineralized after 60 d (Figures 1a and S6) by an anaerobic microbial community. In subsequent discussion, the extent of biodegradation will be expressed relative to (calculated) full mineralization of the sample, unless otherwise noted. Among the three esterified CNFs, the hexyl esterified CNF (DSoverall: 0.09) exhibited a biodegradation rate comparable to unmodified CNF, while dodecyl (DSoverall: 0.45) and phenyl (DSoverall: 0.14) esterified CNFs displayed considerably slower biodegradation rates (Figure 1b), although all three esters were almost completely biodegraded (> 90%) over 424 d (Eq. 1, Table S5). TEMPO-oxidized CNFs containing carboxylate groups with Na<sup>+</sup> and H<sup>+</sup> (Figure S6)

counterions (DSoverall: 0.24) also biodegraded at markedly slower rates, but ultimately underwent almost complete biodegradation analogous to the behavior of unmodified CNFs. In contrast, etherified nanocellulose was dramatically less susceptible to mineralization even at extremely low DSoverall values, with hexyl (DSoverall: 0.05) and dodecyl (DSoverall: 0.11) etherified CNFs only biodegrading to 15% and 10% of the levels exhibited by unmodified CNF, respectively, after 424 d of incubation. Biodegradation of functionalized CNFs by an aerobic microbial community, as found in aerobic wastewater, for 60 d revealed similar trends of functional group-induced inhibition towards biodegradation (Figure 1c, 1d), with unmodified CNFs exhibiting 80 % mass loss, hexyl and phenyl CNF esters and carboxylated CNFs all exhibiting mass loss in excess of 9%, and etherified CNFs exhibiting no measurable mass loss.

In the present study, we have measured the effect of different functionalization strategies on the biodegradation of nanocellulose exposed to the same microbial community, revealing that CNF ethers become non-biodegradable at low DS values (≈0.1), in contrast to the behavior of CNF esters at similar DS values. Qualitatively, the trends observed in the relative inhibition of biodegradation induced by introducing specific functional groups mirror those observed for macrocellulose. For example, previous studies on cellulose functionalized with carboxyl groups have also shown over 50% sample biodegradation in soil burial tests. <sup>88</sup> This behavior that has been attributed to the increased swelling of the cellulosic fiber that occurs upon addition of the hydrophilic carboxylic acid functional groups which facilitates enzymatic ingress into the interior of the cellulosic material. <sup>88, 89</sup> The biodegradability of esterified cellulose observed in this study has also been observed for macrocellulose and has been previously attributed to the susceptibility of ester linkages to enzymatic hydrolysis which regenerates the glucose monomer unit of cellulose. <sup>32, 39, 40, 90</sup> The recalcitrance of ether linkages to biodegradation observed in Figure 1 has

been attributed to their resistance to enzymatic attack/hydrolysis.  $^{32, 41, 91}$  Indeed, we observed a complete lack of biogas production from ethyl cellulose, a commercial, non-biodegradable, macrocellulose ether that produced no biogas over 424 d (Figure 1a). This recalcitrance to mineralization was found for etherified CNFs with very small DS<sub>overall</sub> values ( $\approx 0.1$ ).

During the course of our experiments in anaerobic media, and in contrast to our expectations, we observed that an esterified CNF with relatively high DSoverall (dodecyl, DSoverall 0.45) biodegraded similarly (Figure 1a) to a CNF ester with significantly lower DSoverall (phenyl, DSoverall 0.14). Based on existing literature for macrocellulose, a threefold increase in DSoverall would be expected to decrease the biodegradability of esterified CNFs by over 90%. These data suggested that conventional DSoverall values may not be predictive of the relative biodegradability of functionalized nanocellulose. One possible explanation is that the large surface area-to-volume ratio of nanocellulose causes the surface of functionalized CNFs to take on increased importance relative to macrocellulose. To explore this possibility, a series of esterified CNFs with varying degrees of surface (DSsurface) and overall (DSoverall) functionalization was synthesized, characterized, and biodegraded by anaerobic microorganisms where comparisons of biodegradation behavior were made easier by virtue of our ability to track biogas formation as a function of incubation time.

CNFs were functionalized with dodecyl ester groups using liquid-phase dodecanoic acid (DA-CNF; Figure 2a-d) and lauroyl chloride (LC-CNF; Figures S7, S8). Elemental analysis revealed that by varying reaction conditions, functionalized CNFs with a range of DS<sub>overall</sub> values (Table S1) could be prepared for both sets of CNF esters (DA-CNF and LC-CNF). ATR-FTIR provided spectroscopic evidence of functionalization through the observation of CH<sub>2</sub> (2920 cm<sup>-1</sup> and 2850 cm<sup>-1</sup>) and C=O (1700 cm<sup>-1</sup>) stretching modes (LC-CNF; Figure S7) in addition to the

characteristic O–H (3339 cm<sup>-1</sup>), C–H (2905 cm<sup>-1</sup>), and C–O (1031 cm<sup>-1</sup>) stretches of cellulose. Moreover, a linear relationship between the DS<sub>overall</sub> values obtained from elemental analysis and the C=O (ester):C–O (cellulose) vibrational band ratio of DA-CNFs was observed (Figure 2c), suggesting that ATR-FTIR can serve as a facile, non-destructive alternative to elemental analysis for determining DS<sub>overall</sub> values of functionalized nanocellulose.

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Solid-state <sup>13</sup>C-NMR qualitatively confirmed the trends in DS<sub>overall</sub> as shown in Figure 2b (increasing for DA-CNF-4, 3, and 2).. The <sup>13</sup>C-NMR spectrum of unmodified CNF consists of peaks between 50 and 110 ppm arising from carbons 1-6 in cellulose (C1-C6, labeled in Figure S9 and Table S6) and includes peaks arising from amorphous and crystalline nanocellulose. In addition to these principal cellulose peaks, the spectra of esterified CNF contain ester (180 ppm) and methylene (32, 25, 15 ppm) peaks that increased in intensity with increasing reaction time (in order of DA-CNF sample 1, 4, 3, 2). The NMR spectra reveal a minor increase in crystallinity (39% to 58%) of the esterified CNF samples compared to the unmodified CNF sample (Table S7). This is not expected to significantly contribute to differences in biodegradation, however, as crystalline and non-crystalline nanocellulose exhibit similar biodegradation properties<sup>92</sup> (Figure S10). We calculated DS<sub>overall</sub> from variable contact time cross polarization—magic-angle spinning experiments for two samples: phenyl ester CNF and DA-CNF-2 (Table S8, Figure S11). Variable contact time experiments and elemental analysis produced the same ranking of DS<sub>overall</sub>. The values of DS<sub>overall</sub> for the phenyl ester CNF samples were more similar for the two measurements than they were for the DA-CNF-2 samples. Differences in DS<sub>overall</sub> derived from variable contact time experiments from those determined by elemental analysis are possibly due to multiple  $T_{1\rho H}$ behavior.

X-ray Photoelectron Spectroscopy was used to determine DS<sub>surface</sub> values. Specifically, the

C(1s) region of unmodified CNF contains C–C, C–O, and O–C–O components with peaks at 285.0 eV, 286.6 eV, and 288.5 eV, <sup>93</sup> respectively (Figures 2d, S8), while the O(1s) region features a single broad peak centered at 533.2 eV. With increased DS<sub>surface</sub>, the C–C component increases in intensity due to the grafting of long alkyl chain ester groups. Importantly, changes in DS<sub>surface</sub> did not correlate with changes in DS<sub>overall</sub>; for example, DA-CNF-2 features the second lowest DS<sub>surface</sub> (0.10), but the highest DS<sub>overall</sub> (0.45). In addition to liquid-phase esterification, gas-phase reactions using lauroyl or hexanoyl chloride were performed with the expectation that this approach would restrict functionalization to the CNF surface (Figure S12). The XPS spectra in Figure S12 confirms that measurable increases in DS<sub>surface</sub> occurred after gas-phase CNF functionalization in the absence of significant bulk functionalization (Figure S13).

Esterified CNFs functionalized with dodecanoic acid (DA-CNFs) and lauroyl chloride (LC-CNFs) were biodegraded by an anaerobic microbial community (Figures 3 and S14) to assess the sensitivity of CNF biodegradability to changes in DS<sub>surface</sub>, DS<sub>overall</sub>, and/or both. The rate and extent of biodegradation of both CNF types were found to change systematically in response to changes in DS<sub>surface</sub>, but not DS<sub>overall</sub> (Figure 3, Table S9, Figure S15). This trend is most clearly demonstrated by DA-CNF samples where increases in DS<sub>surface</sub> led to systematic decreases in biodegradation, while DS<sub>overall</sub> values did not correlate with biodegradation trends (Figure 3a, Figure S15). As an example, DA-CNF-2 possessed the highest DS<sub>overall</sub> value (0.45) of the DA-CNFs, and yet was almost completely biodegraded (94%), albeit at a slower (29%) rate compared to unmodified CNFs.

Analogous behavior is observed with LC-CNFs. As the extent of surface functionalization (DS<sub>surface</sub>) increased in LC-CNF samples, the extent and rate of biogas production decreased, with the most extensively surface functionalized CNF in this series (LC-CNF-4) exhibiting a biogas

production rate and extent of biodegradation only 24% and 37% of unmodified CNF, respectively (Figure 3b). The lack of correlation with DS<sub>overall</sub> is observed most clearly in LC-CNF-3: this sample featured the lowest DS<sub>overall</sub> (0.56) of the LC-CNFs but was more recalcitrant to biodegradation than LC-CNF-1 and LC-CNF-2, which each featured a DS<sub>overall</sub> of approximately 0.8 (Figure 3b).

We note that LC-CNF samples experienced more extensive overall biodegradation than DA-CNF samples despite LC-CNF samples reaching higher DS<sub>surface</sub> values (max 2.46) than DA-CNFs (max 0.43). This behavior is a direct result of the production of HCl during CNF functionalization with acyl chlorides, which reduces cellulose chain length and particle size.<sup>63, 94-97</sup> This decrease in chain length increases the overall biodegradability of cellulosic materials by offering more sites/surface area (e.g., chain ends) for the initiation of enzymatic attack.<sup>55, 98, 99</sup> In contrast, esterification using carboxylic acid reagents as used in the synthesis of DA-CNFs does not produce HCl at the site of functionalization, limiting damage to the cellulose chain, thereby producing DA-CNF samples with lower DS<sub>surface</sub> values which undergo a smaller degree of biodegradation.

Gas-phase functionalization was used to specifically target the role of surface functionalization in biodegradation. To this end, CNF surfaces were modified with lauroyl chloride (GP-LC-CNF) and hexanoyl chloride (GP-HC-CNF). Hexanoyl chloride enabled a wider range of and higher DS<sub>surface</sub> values to be achieved (1.19-2.43) compared to lauroyl chloride (0.07-0.33) due to its higher volatility (hexanoyl chloride  $T_b \approx 150$  °C vs. lauroyl chloride  $T_b \approx 260$  °C). Because these esterified samples exhibited levels of overall substitution (i.e., DS<sub>overall</sub> < 0.17, Figure S13) that would not slow the biodegradation of CNF esters (see Table S9), the effect of surface functionalization on the biodegradation of nanocellulose could be isolated (Figures 3c-d and S14).

The extremely low levels of DS<sub>overall</sub> produced by gas-phase functionalization compared to the corresponding DS<sub>surface</sub> values can be appreciated if we consider that the CNFs have diameters  $\sim 50$  nm and that the XPS measurements of the near-surface region are dominated by photoelectrons from the outermost 3 nm of the CNFs. <sup>100</sup> Thus, even in the event that 100% of the CNF hydroxyl groups in the near-surface region were functionalized (i.e., DS<sub>surface</sub> of 3.0), the corresponding DS<sub>overall</sub> would be only approximately 0.2.

The specificity of surface functionalization realized with these gas-phase modified CNF samples provided a clear indication of the role that DS<sub>surface</sub> plays in regulating CNF biodegradability. As seen in Figure 3c, GP-HC-CNF-1, the least functionalized GP-HC-CNF samples (DS<sub>surface</sub> 1.19), displayed a 60% reduction in biogas production rate compared to unmodified CNF. In contrast, the most surface functionalized GP-HC-CNF (GP-HC-CNF-4; DS<sub>surface</sub> 2.43) exhibited a biogas production rate only 17% of that observed for unmodified CNF. The extent of biodegradation followed the same dependence on DS<sub>surface</sub>, with GP-HC-CNF-1 and GP-HC-CNF-4 samples undergoing 100% and 70% biodegradation, respectively. As expected, the CNFs functionalized with gas phase lauroyl chloride were also more recalcitrant to biodegradation (Figure 3d) than unmodified CNF, but less so than GP-HC-CNFs (lower DS<sub>surface</sub> values). GP-LC-CNF samples with DS<sub>surface</sub> values > 0.17 exhibited measurable decreases in the biogas production rate (> 40% reduction), although the extent of biodegradation reached approximately 90% of the value expected for unmodified CNFs after 75 d due to the relatively low (< 0.35 DS<sub>surface</sub>) levels of surface functionalization.

Figure 3 reveals that for both gas- and liquid-phase CNF functionalization, the extent and rate of biodegradation decrease with increasing DS<sub>surface</sub> values, largely independent of DS<sub>overall</sub> values. This dependency on surface chemistry is consistent with the environmental properties of

other carbon nanomaterials such as carbon nanotubes, <sup>101, 102</sup> carbon dots, <sup>103-105</sup> and graphene, <sup>106, 107</sup> which have previously been found to be influenced—and in some cases wholly determined—by surface properties. Of significance, the observed reduction in biodegradability for functionalized nanocellulose manifested as a decreased propensity to be mineralized into biogas (i.e., complete biodegradation). Consequently, either the parent material or products of incomplete biodegradation may persist even in conditions with high microbial activity and result in environmental accumulation.

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The importance of surface functionalization is likely a reflection of the biodegradation mechanism of cellulosic materials, which is typically initiated at the surface via highly specific interactions with microbial enzymes.<sup>26, 56, 59</sup> For example, during anaerobic biodegradation, cellulose is completely mineralized by microorganisms which initiate the process using extracellular cellulosomes or multiprotein complexes of cellulolytic enzymes.<sup>25-27</sup> These enzymes are particularly sensitive to the surface of the substrate material, and microorganisms alter the structural and enzymatic composition of the cellulosome to suit the substrate in question.<sup>57</sup> The small length of the glucose subunits of cellulose (roughly 0.5 nm)<sup>108</sup> compared to that of cellulosomes (roughly 18 nm)<sup>109</sup> suggests that functional groups covalently attached to surface subunits of nanocellulose, even at low concentrations, must be removed before conventional enzymatic degradation can proceed. 57,58,110 Specifically, biodegradation will be delayed until these functionalized surface sites have been sufficiently removed to regenerate a cellulose substrate recognizable to microbial cellulosomes. The surface-dependent hydrolysis of these functionalized sites serves as the rate-limiting step for the biodegradation of functionalized nanocellulose and likely explains why the bulk of the material is of less importance in determining the rate and extent of biodegradation.

For esterified CNFs with relatively low degrees of surface functionalization (e.g., DA-CNF-1, phenyl ester CNF;  $DS_{overall} \le 0.14$ ), the presence of ester groups at the surface causes a decrease in biodegradation rate—although the CNF still ultimately biodegrades. However, as the degree of surface functionalization increases (e.g., DA-CNF-4, LC-CNF-4), our data indicates that an increasing fraction of the CNFs are recalcitrant to biodegradation (see Figure 3) despite the enzymatic susceptibility of ester groups. We ascribe this effect to the concentrations of ester groups in certain regions of the surface being sufficiently high to interfere with enzyme regioselectivity, blocking esterases from properly orienting with a single ester group and thus preventing their hydrolysis. 90 As the DS value increases, the fraction of the CNF surface covered with sufficiently high concentrations of ester groups to prevent biodegradation increases. This argument is supported by the observation that esterases are unable to biodegrade macrocellulose esters with high DS values<sup>31, 48, 90</sup> The observed inhibitory effect is enhanced when the covalent linkages are resistant to enzymatic hydrolysis, as is the case for etherified CNFs, 91 where a DS<sub>surface</sub> of 0.16 and 0.25 was sufficient to prevent biodegradation of dodecyl and hexyl ether CNFs, respectively (see Figure 1). Indeed, the larger DS<sub>surface</sub> vs DS<sub>overall</sub> values for etherified CNFs (e.g., 0.16 vs. 0.11, respectively for dodecyl ether CNF; see Figure 1b) helps to explain why these functionalized CNFs were so recalcitrant to biodegradation even at low levels of DS<sub>overall</sub>.

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In contrast to our findings, DS<sub>overall</sub> is generally found to be a reasonable predictor for the biodegradation behavior of macrocellulose, with the degree of surface functionalization rarely reported in biodegradation studies.<sup>31, 46-48</sup> One potential explanation for this difference between macro- and nanocellulose is that a stronger correlation between DS<sub>overall</sub> and DS<sub>surface</sub> may exist for macrocellulose. In this respect, we note that macrocellulose exhibits an increased swelling capacity (~48% vs. ~26% for macrocellulose compared to CNFs in aqueous media). This will almost

certainly increase the ingress of chemical reagents into the interior of macrocellulose during liquid phase functionalization, likely leading to more similar degrees of bulk vs surface functionalization in macrocellulose compared to CNFs. 111-113 Regardless of the detailed mechanistic underpinnings, results from this investigation highlight the need to measure both DSoverall and DSsurface for functionalized macrocellulose, and to establish the influence of these two metrics on biodegradation properties.

### **Implications**

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The increased recalcitrance of surface functionalized nanocellulose to mineralization is undesirable because complete biodegradation is necessary to ensure its removal from the environment, minimizing accumulation and any consequent impact.<sup>23, 24</sup> The present study has revealed that the extent of surface functionalization and type of covalent linkage strongly influence the degree of CNF recalcitrance to biodegradation. We note that the use of microbial communities derived from wastewater represent optimized conditions for the biodegradation of cellulose due to the diversity, activity, and concentration of microorganisms in the culture as well as the availability of nutrients in the BMP tests. 16, 114-116 Therefore, under environmental conditions where less diverse microbial communities exist (e.g., soils, aquatic sediments), the effect of functionalization on nanocellulose mineralization is expected to be more pronounced relative to the effects observed in this study. We note that the primary biological transformation products of functionalized nanocellulose are expected be carbon dioxide and methane in anaerobic environments and carbon dioxide in aerobic environments; both gases contribute to the greenhouse effect. Thus, although biodegradation is typically viewed as a positive environmental outcome because it acts to remove otherwise persistent materials its effects are not without consequences.

We found that although relative biodegradation trends among different functionalized CNFs were independent of the microbial community (i.e., aerobic vs. anaerobic), the magnitude of inhibition differed (Figure 1d). Specifically, the extent of biodegradation was reduced in aerobic wastewater compared to anaerobic wastewater, likely due to differences in microbial population and numbers. The decreased aerobic biodegradation of functionalized CNF suggests that anaerobic digestion should be utilized to maximize biodegradation of functionalized CNFs and reduce landfill disposal.

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While sustainability has been identified as an area of focus in the production phase of surface-modified nanocellulose, <sup>64, 73, 117</sup> the end-of-life environmental fate of such nanomaterials has been largely overlooked or assumed to be comparable to unmodified nanocellulose. 1, 21, 118-120 Importantly, products which utilize surface-functionalized nanocellulose and are marketed as biodegradable (e.g., packaging materials)<sup>21, 121</sup> may actually feature environmentally persistent nanocellulose. For example, the combination of surface-esterified nanocellulose with the biodegradable polymer poly(3-hydroxybutyrate-co-3-hydroxyvalerate) (PHBV) to create a strengthened material deemed appropriate for use as a fully biodegradable food packaging material.<sup>21</sup> However, based on our data, the surface-level esterification of nanocellulose used in the reinforcement of PHBV would significantly reduce its overall biodegradability. The same argument holds true for functionalized nanocellulose used in other applications such as in displays and coatings. 12, 122-125 For example, Granström et al., developed a stearoyl ester CNF-based aerogel with projected applications in coatings and insulators<sup>12</sup> that our study indicates will not biodegrade as rapidly as unmodified nanocellulose. Indeed, applications of esterified CNFs in packaging materials, coatings, and lubricants are expected to spur a growth in production to meet increased demand.<sup>21, 62, 121, 122, 126</sup> Our study highlights that the commercial benefit achieved through functionalization of nanocellulose must be carefully weighed against the consequent changes in the persistence of these nanomaterials. For example, decreasing the cellulose chain length and DS for CNFs functionalized with esters can be anticipated to increase biodegradability, but by the same token these changes are also likely to decrease CNF dispersibility in organic solvents with potential impacts on materials properties. Another practical consequence of the findings from this study is that even relatively low degrees of CNF surface functionalization lead to a portion of the material becoming recalcitrant to biodegradation. Moreover, due to the differential influence of DS<sub>surface</sub> and DS<sub>overall</sub> on the biodegradation of functionalized CNF direct comparisons of the effects that different functional groups play in mediating CNF biodegradation is difficult because both DS<sub>overall</sub> and DS<sub>surface</sub> need to be similar to isolate the impact of different functional groups on the biodegradation of the functionalized CNF. However, due to the difference in bulk vs. surface accessibility and reagent reactivity, exerting control over these two parameters experimentally is difficult.

### **Supporting Information**

Detailed information about instrumental parameters for material characterization (CHN analysis, ATR-FTIR, XPS, NMR). Specific information on solid state NMR experiments, theory, and data (e.g., crystallinity and peak positions). Tables listing DSoverall and DSsurface values with corresponding CHN and XPS data for all samples. Comparison of calculated maximum biogas production to empirical biogas production volume. Biogas production curves for CNF and functionalization reagents which enabled comparison of biodegradation from different functionalized CNFs. Gompertz model parameters for kinetic modeling of biogas production curves. Complete ATR-FTIR and XPS spectra for samples examined in the study. Unnormalized

biogas production curves for functionalized CNFs. Direct comparison of DS<sub>surface</sub> and DS<sub>overall</sub> with maximum biogas production from each sample.

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#### References

- 645 1. Kalia, S.; Dufresne, A.; Cherian, B. M.; Kaith, B. S.; Averous, L.; Njuguna, J.; Nassiopoulos, E.,
- 646 Cellulose-Based Bio- and Nanocomposites: A Review. Int. J. Polym. Sci. 2011, 2011.
- 2. Dufresne, A., *Nanocellulose: From Nature to High Performance Tailored Materials*. Walter de Gruyter: Berlin, Germany, 2012.
- Hamid, S. B. A.; Zain, S. K.; Das, R.; Centi, G., Synergic Effect of Tungstophosphoric Acid and
- 650 Sonication for Rapid Synthesis of Crystalline Nanocellulose. Carbohydr. Polym. 2016, 138, 349-355.
- 4. Phanthong, P.; Karnjanakom, S.; Reubroycharoen, P.; Hao, X.; Abudula, A.; Guan, G., A Facile
- One-Step Way for Extraction of Nanocellulose with High Yield by Ball Milling with Ionic Liquid.
- 653 *Cellulose* **2017**, *24*, (5), 2083-2093.
- 654 5. Henriksson, M.; Henriksson, G.; Berglund, L. A.; Lindström, T., An Environmentally Friendly
- Method for Enzyme-Assisted Preparation of Microfibrillated Cellulose (MFC) Nanofibers. *Eur. Polym. J.* **2007**, *43*, (8), 3434-3441.
- 657 6. Moon, R. J.; Martini, A.; Nairn, J.; Simonsen, J.; Youngblood, J., Cellulose nanomaterials
- review: structure, properties and nanocomposites. Chem. Soc. Rev. 2011, 40, (7), 3941-3994.
- 659 7. Bharimalla, A. K.; Deshmukh, S. P.; Vigneshwaran, N.; Patil, P. G.; Prasad, V., Nanocellulose-
- Polymer Composites for Applications in Food Packaging: Current Status, Future Prospects and
- 661 Challenges. *Polym. Plast. Technol. Eng.* **2017**, *56*, (8), 805-823.
- Wei, H.; Rodriguez, K.; Renneckar, S.; Vikesland, P. J., Environmental science and engineering
- applications of nanocellulose-based nanocomposites. *Environ. Sci.: Nano* **2014**, *1*, (4), 302-316.
- 664 9. Li, Y.-Y.; Wang, B.; Ma, M.-G.; Wang, B., Review of Recent Development on Preparation,
- Properties, and Applications of Cellulose-Based Functional Materials. Int. J. Polym. Sci. 2018.
- 666 10. Ummartyotin, S.; Manuspiya, H., A critical review on cellulose: From fundamental to an
- approach on sensor technology. *Renewable and Sustainable Energy Reviews* **2015**, *41*, 402-412.
- Mohammed, N.; Grishkewich, N.; Tam, K. C., Cellulose nanomaterials: promising sustainable
- nanomaterials for application in water/wastewater treatment processes. *Environ. Sci.: Nano* **2018**, *5*, (3), 623-658.
- 671 12. Granström, M.; née Pääkkö, M. K.; Jin, H.; Kolehmainen, E.; Kilpeläinen, I.; Ikkala, O., Highly
- water repellent aerogels based on cellulose stearoyl esters. *Polymer Chemistry* **2011**, *2*, (8), 1789-1796.
- 673 13. Phanthong, P.; Reubroycharoen, P.; Hao, X.; Xu, G.; Abudula, A.; Guan, G., Nanocellulose:
- Extraction and application. Carbon Resources Conversion 2018, 1, (1), 32-43.
- 675 14. Qiu, X.; Hu, S., "Smart" Materials Based on Cellulose: A Review of the Preparations, Properties,
- and Applications. *Materials* **2013**, *6*, (3), 738-781.
- Nasseri, R.; Deutschman, C. P.; Han, L.; Pope, M. A.; Tam, K. C., Cellulose nanocrystals in
- smart and stimuli-responsive materials: a review. *Materials Today Advances* **2020**, *5*, 100055.
- 679 16. Frank, B. P.; Durkin, D. P.; Caudill, E. R.; Zhu, L.; White, D. H.; Curry, M. L.; Pedersen, J. A.;
- Fairbrother, D. H., Impact of Silanization on the Structure, Dispersion Properties, and Biodegradability of
- Nanocellulose as a Nanocomposite Filler. ACS Applied Nano Materials 2018, 1, (12), 7025-7038.
- 682 17. Nakatani, H.; Iwakura, K.; Miyazaki, K.; Okazaki, N.; Terano, M., Effect of Chemical Structure
- of Silane Coupling Agent on Interface Adhesion Properties of Syndiotactic Polypropylene/Cellulose
- 684 Composite. J. Appl. Polym. Sci. **2011**, 119, (3), 1732-1741.
- 685 18. Chin, K.-M.; Sung Ting, S.; Ong, H. L.; Omar, M., Surface functionalized nanocellulose as a
- veritable inclusionary material in contemporary bioinspired applications: A review. J. Appl. Polym. Sci.
- **2018**, *135*, (13), 46065.
- 688 19. Brun, V.; Hansen, F.; Turpin, D.; Bennett, K., Cellulose Nanomaterials Research Roadmap.
- 689 Agenda 2020 Technology Alliance 2016.
- 690 20. Madsen, L. D.; Svedberg, E. B., Materials Research for Manufacturing: An Industrial
- 691 Perspective of Turning Materials into New Products. Springer International Publishing: Switzerland,
- 692 2016; Vol. 224.

- 693 21. Yu, H.; Yan, C.; Yao, J., Fully biodegradable food packaging materials based on functionalized
- 694 cellulose nanocrystals/poly(3-hydroxybutyrate-co-3-hydroxyvalerate) nanocomposites. RSC Adv. 2014, 4,
- 695 (104), 59792-59802.
- 696 22. Carpenter, A. W.; de Lannoy, C. F.; Wiesner, M. R., Cellulose Nanomaterials in Water Treatment
- 697 Technologies. Environ. Sci. Technol. 2015, 49, (9), 5277-5287.
- 698 23. Zumstein, M. T.; Narayan, R.; Kohler, H.-P. E.; McNeill, K.; Sander, M., Dos and Do Nots When
- Assessing the Biodegradation of Plastics. *Environ. Sci. Technol.* **2019**, *53*, (17), 9967-9969.
- 700 24. Zumstein, M. T.; Schintlmeister, A.; Nelson, T. F.; Baumgartner, R.; Woebken, D.; Wagner, M.;
- Kohler, H.-P. E.; McNeill, K.; Sander, M., Biodegradation of synthetic polymers in soils: Tracking
- carbon into CO(2) and microbial biomass. Sci Adv 2018, 4, (7), eaas 9024-eaas 9024.
- 703 25. Pérez, J.; Muñoz-Dorado, J.; de la Rubia, T.; Martínez, J., Biodegradation and biological
- treatments of cellulose, hemicellulose and lignin: an overview. *International Microbiology* **2002**, *5*, (2), 53-63.
- 706 26. Leschine, S. B., Cellulose Degradation in Anaerobic Environments. *Annu. Rev. Microbiol.* **1995**, 707 49, (1), 399-426.
- 708 27. Béguin, P.; Aubert, J.-P., The biological degradation of cellulose. *FEMS Microbiology Reviews* 709 **1994**, *13*, (1), 25-58.
- 710 28. Filer, J.; Ding, H. H.; Chang, S., Biochemical Methane Potential (BMP) Assay Method for
- Anaerobic Digestion Research. Water 2019, 11, (5), 921.
- 712 29. Singh, G. Biodegradation of Nanocellulose and Microbial Community Response: Effect of
- 713 Surface Modification and Morphology. Virginia Polytechnic Institute and State University, Blacksburg,
- 714 VA, 2015.
- 715 30. Singh, G.; Chandoha-Lee, C.; Zhang, W.; Renneckar, S.; Vikesland, P. J.; Pruden, A.,
- 716 Biodegradation of Nanocrystalline Cellulose by Two Environmentally-Relevant Consortia. *Water Res.*
- **2016,** *104*, 137-146.
- 718 31. Puls, J.; Wilson, S. A.; Hölter, D., Degradation of Cellulose Acetate-Based Materials: A Review.
- 719 *Journal of Polymers and the Environment* **2011**, *19*, (1), 152-165.
- 720 32. Reese, E. T., Biological Degradation of Cellulose Derivatives. *Industrial & Engineering*
- 721 *Chemistry* **1957**, *49*, (1), 89-93.
- 33. Sakai, K.; Yamauchi, T.; Nakasu, F.; Ohe, T., Biodegradation of Cellulose Acetate by Neisseria
- 723 sicca. Bioscience, Biotechnology, and Biochemistry 1996, 60, (10), 1617-1622.
- 724 34. Haider, T. P.; Völker, C.; Kramm, J.; Landfester, K.; Wurm, F. R., Plastics of the Future? The
- 725 Impact of Biodegradable Polymers on the Environment and on Society. *Angewandte Chemie*
- 726 International Edition **2019**, *58*, (1), 50-62.
- 727 35. Ly, B.; Thielemans, W.; Dufresne, A.; Chaussy, D.; Belgacem, M. N., Surface functionalization
- of cellulose fibres and their incorporation in renewable polymeric matrices. *Compos. Sci. Technol.* **2008**,
- 729 *68*, (15), 3193-3201.
- 730 36. Jandura, P.; Kokta, B. V.; Riedl, B., Cellulose Fibers/Polyethylene Hybrid Composites: Effect of
- 731 Long Chain Organic Acid Cellulose Esters and Organic Peroxide on Rheology and Tensile Properties.
- *Journal of Reinforced Plastics and Composites* **2001**, *20*, (8), 697-717.
- 733 37. de Carvalho Oliveira, G.; Filho, G. R.; Vieira, J. G.; De Assunção, R. M. N.; da Silva Meireles,
- 734 C.; Cerqueira, D. A.; de Oliveira, R. J.; Silva, W. G.; de Castro Motta, L. A., Synthesis and application of
- 735 methylcellulose extracted from waste newspaper in CPV-ARI Portland cement mortars. J. Appl. Polym.
- 736 *Sci.* **2010,** *118*, (3), 1380-1385.
- 737 38. Abushammala, H.; Mao, J., A Review of the Surface Modification of Cellulose and
- Nanocellulose Using Aliphatic and Aromatic Mono- and Di-Isocyanates. *Molecules* **2019**, *24*, (15), 2782.
- 739 39. Jianlong, W.; Lujun, C.; Hanchang, S.; Yi, Q., Microbial degradation of phthalic acid esters under
- 740 anaerobic digestion of sludge. *Chemosphere* **2000**, *41*, (8), 1245-1248.
- 741 40. Liu, S.; Suflita, J. M., Anaerobic biodegradation of methyl esters by Acetobacterium woodii and
- Eubacterium limosum. *Journal of Industrial Microbiology* **1994**, *13*, (5), 321-327.

- 743 41. Reese, E. T.; Siu, R. G. H.; Levinson, H. S., The biological degradation of soluble cellulose
- derivatives and its relationship to the mechanism of cellulose hydrolysis. *Journal of bacteriology* **1950**,
- 745 *59*, (4), 485-497.
- 746 42. Vaca-Garcia, C.; Borredon, M. E.; Gaseta, A., Determination of the degree of substitution (DS) of
- mixed cellulose esters by elemental analysis. *Cellulose* **2001**, 8, (3), 225-231.
- 748 43. Samaranayake, G.; Glasser, W. G., Cellulose derivatives with low DS. I. A novel acylation
- 749 system. *Carbohydr. Polym.* **1993,** *22*, (1), 1-7.
- 750 44. King, A. W. T.; Jalomäki, J.; Granström, M.; Argyropoulos, D. S.; Heikkinen, S.; Kilpeläinen, I.,
- A new method for rapid degree of substitution and purity determination of chloroform-soluble cellulose
- 752 esters, using 31P NMR. Anal. Methods **2010**, 2, (10), 1499-1505.
- 753 45. Blanchard, F. A.; Takahashi, I. T.; Alexander, H. C., Biodegradability of [14C]methylcellulose by
- 754 activated sludge. *Appl. Environ. Microbiol.* **1976,** *32*, (4), 557-560.
- 755 46. Wirick, M. G., A study of the enzymic degradation of CMC and other cellulose ethers. *Journal of*
- 756 *Polymer Science Part A-1: Polymer Chemistry* **1968**, *6*, (7), 1965-1974.
- 757 47. Wirick, M. G., Aerobic Biodegradation of Carboxymethylcellulose. *Journal (Water Pollution*
- 758 *Control Federation*) **1974**, *46*, (3), 512-521.
- 759 48. Rivard, C. J.; Adney, W. S.; Himmel, M. E.; Mitchell, D. J.; Vinzant, T. B.; Grohmann, K.;
- Moens, L.; Chum, H., Effects of natural polymer acetylation on the anaerobic bioconversion to methane
- and carbon dioxide. *Applied Biochemistry and Biotechnology* **1992**, *34*, (1), 725-736.
- 762 49. Glasser, W. G.; McCartney, B. K.; Samaranayake, G., Cellulose derivatives with a low degree of
- substitution. 3. The biodegradability of cellulose esters using a simple enzyme assay. *Biotechnology*
- 764 *Progress* **1994,** 10, (2), 214-219.
- 765 50. Owen, W. F.; Stuckey, D. C.; Healy, J. B.; Young, L. Y.; McCarty, P. L., Bioassay for
- Monitoring Biochemical Methane Potential and Anaerobic Toxicity. Water Res. 1979, 13, (6), 485-492.
- 767 51. Hayakawa, C.; Funakawa, S.; Fujii, K.; Kadono, A.; Kosaki, T., Effects of climatic and soil
- 768 properties on cellulose decomposition rates in temperate and tropical forests. Biology and Fertility of
- 769 *Soils* **2014,** *50*, (4), 633-643.
- 770 52. Hofsten, B. V.; Edberg, N., Estimating the Rate of Degradation of Cellulose Fibers in Water.
- 771 *Oikos* **1972,** *23*, (1), 29-34.
- 772 53. Goodwin, D. G.; Marsh, K. M.; Sosa, I. B.; Payne, J. B.; Gorham, J. M.; Bouwer, E. J.;
- 773 Fairbrother, D. H., Interactions of Microorganisms with Polymer Nanocomposite Surfaces Containing
- 774 Oxidized Carbon Nanotubes. *Environ. Sci. Technol.* **2015**, *49*, (9), 5484-5492.
- 775 54. Zhao, X.; Cornish, K.; Vodovotz, Y., Narrowing the Gap for Bioplastic Use in Food Packaging:
- 776 An Update. Environ. Sci. Technol. 2020.
- 777 55. Chinaglia, S.; Tosin, M.; Degli-Innocenti, F., Biodegradation rate of biodegradable plastics at
- 778 molecular level. *Polym. Degrad. Stab* **2018**, *147*, 237-244.
- 779 56. Tokiwa, Y.; Calabia, B. P.; Ugwu, C. U.; Aiba, S., Biodegradability of plastics. Int J Mol Sci
- **2009,** *10*, (9), 3722-3742.
- 781 57. Artzi, L.; Bayer, E. A.; Moraïs, S., Cellulosomes: bacterial nanomachines for dismantling plant
- 782 polysaccharides. *Nature Reviews Microbiology* **2017**, *15*, (2), 83-95.
- 783 58. Fierobe, H.-P.; Bayer, E. A.; Tardif, C.; Czjzek, M.; Mechaly, A.; Bélaïch, A.; Lamed, R.;
- Shoham, Y.; Bélaïch, J.-P., Degradation of Cellulose Substrates by Cellulosome Chimeras: Substrate
- 785 Targeting Versus Proximity Of Enzyme Components. Journal of Biological Chemistry 2002, 277, (51),
- 786 49621-49630.
- 787 59. Wang, Z.-W.; Lee, S.-H.; Elkins, J. G.; Morrell-Falvey, J. L., Spatial and temporal dynamics of
- 788 cellulose degradation and biofilm formation by Caldicellulosiruptor obsidiansis and Clostridium
- 789 thermocellum. *AMB Express* **2011**, *1*, 30-30.
- 790 60. Ioelovich, M., Cellulose as a Nanostructured Polymer: A Short Review. 2008 2008, 3, (4), 16.
- 791 61. Ioelovich, M., Characterization of Various Kinds of Nanocellulose. In *Handbook of*
- *Nanocellulose and Cellulose Nanocomposites*, 2017; pp 51-100.

- Wang, Y.; Wang, X.; Xie, Y.; Zhang, K., Functional nanomaterials through esterification of cellulose: a review of chemistry and application. *Cellulose* **2018**, *25*, (7), 3703-3731.
- 795 63. Willberg-Keyriläinen, P.; Ropponen, J., Evaluation of esterification routes for long chain cellulose esters. *Heliyon* **2019**, *5*, (11), e02898.
- 797 64. Berlioz, S.; Molina-Boisseau, S.; Nishiyama, Y.; Heux, L., Gas-Phase Surface Esterification of Cellulose Microfibrils and Whiskers. *Biomacromolecules* **2009**, *10*, (8), 2144-2151.
- 799 65. Fumagalli, M.; Sanchez, F.; Boisseau, S. M.; Heux, L., Gas-phase esterification of cellulose
- nanocrystal aerogels for colloidal dispersion in apolar solvents. *Soft Matter* **2013**, *9*, (47), 11309-11317.
- 801 66. David, G.; Gontard, N.; Guerin, D.; Heux, L.; Lecomte, J.; Molina-Boisseau, S.; Angellier-
- Coussy, H., Exploring the potential of gas-phase esterification to hydrophobize the surface of micrometric cellulose particles. *Eur. Polym. J.* **2019**, *115*, 138-146.
- 804 67. Eyley, S.; Thielemans, W., Surface modification of cellulose nanocrystals. *Nanoscale* **2014**, *6*, 805 (14), 7764-7779.
- 806 68. Peydecastaing, J.; Vaca-Garcia, C.; Borredon, E., Accurate determination of the degree of substitution of long chain cellulose esters. *Cellulose* **2008**, *16*, (2), 289.
- 808 69. Söyler, Z.; Meier, M. A. R., Sustainable functionalization of cellulose and starch with diallyl carbonate in ionic liquids. *Green Chem.* **2017**, *19*, (16), 3899-3907.
- 810 70. Zheng, Y.; Song, J.; Cheng, B.; Fang, X.; Yuan, Y., Preparation and flame retardancy of 3-
- (hydroxyphenylphosphinyl)-propanoic acid esters of cellulose and their fibers. *Cellulose* **2015**, *22*, (1), 229-244.
- 813 71. Willberg-Keyriläinen, P.; Orelma, H.; Ropponen, J., Injection Molding of Thermoplastic
- Cellulose Esters and Their Compatibility with Poly(Lactic Acid) and Polyethylene. *Materials (Basel)* **2018**, *11*, (12), 2358.
- 816 72. Son, D.; Cho, S.; Nam, J.; Lee, H.; Kim, M., X-ray-Based Spectroscopic Techniques for
- 817 Characterization of Polymer Nanocomposite Materials at a Molecular Level. *Polymers* **2020**, *12*, (5),
- 818 1053.
- 819 73. Espino-Perez, E.; Domenek, S.; Belgacem, N.; Sillard, C.; Bras, J., Green Process for Chemical
- Functionalization of Nanocellulose with Carboxylic Acids. *Biomacromolecules* **2014**, *15*, (12), 4551-
- 821 4560.
- 822 74. Langley, L. A.; Villanueva, D. E.; Fairbrother, D. H., Quantification of Surface Oxides on
- 823 Carbonaceous Materials. *Chemistry of Materials* **2006**, *18*, (1), 169-178.
- Shim, S. H.; Kim, K. T.; Lee, J. U.; Jo, W. H., Facile Method to Functionalize Graphene Oxide
- and Its Application to Poly(ethylene terephthalate)/Graphene Composite. ACS Appl. Mater. Interfaces
- **2012**, *4*, (8), 4184-4191.
- 827 76. Singh, M.; Kaushik, A.; Ahuja, D., Surface functionalization of nanofibrillated cellulose extracted
- from wheat straw: Effect of process parameters. *Carbohydr. Polym.* **2016**, *150*, 48-56.
- 829 77. Braun, E. I.; Pantano, P., The importance of an extensive elemental analysis of single-walled carbon nanotube soot. *Carbon* **2014**, 77, 912-919.
- 831 78. Bohutskyi, P.; Betenbaugh, M. J.; Bouwer, E. J., The effects of alternative pretreatment strategies
- on anaerobic digestion and methane production from different algal strains. *Bioresource Technology*
- **2014**, *155*, 366-372.
- 834 79. Dai, X.; Chen, Y.; Zhang, D.; Yi, J., High-solid Anaerobic Co-digestion of Sewage Sludge and
- 835 Cattle Manure: The Effects of Volatile Solid Ratio and pH. Sci. Rep. 2016, 6, (1), 35194.
- 836 80. Mei, R.; Narihiro, T.; Nobu, M. K.; Kuroda, K.; Liu, W.-T., Evaluating digestion efficiency in
- full-scale anaerobic digesters by identifying active microbial populations through the lens of microbial
- 838 activity. Sci. Rep. **2016**, 6, (1), 34090.
- 839 81. Yi, J.; Dong, B.; Jin, J.; Dai, X., Effect of increasing total solids contents on anaerobic digestion
- of food waste under mesophilic conditions: performance and microbial characteristics analysis. *PloS one*
- **2014,** *9*, (7), e102548-e102548.
- 82. Koch, K.; Hafner, S. D.; Astals, S.; Weinrich, S., Evaluation of Common Supermarket Products
- as Positive Controls in Biochemical Methane Potential (BMP) Tests. Water 2020, 12, (5), 1223.

- 83. Bohutskyi, P.; Keller, T. A.; Phan, D.; Parris, M. L.; Li, M.; Richardson, L.; Kopachevsky, A. M.,
- 845 Co-digestion of Wastewater-Grown Filamentous Algae With Sewage Sludge Improves Biomethane
- Production and Energy Balance Compared to Thermal, Chemical, or Thermochemical Pretreatments.
- 847 *Frontiers in Energy Research* **2019**, *7*, (47).
- 848 84. Li, P.; Li, W.; Sun, M.; Xu, X.; Zhang, B.; Sun, Y., Evaluation of Biochemical Methane Potential
- and Kinetics on the Anaerobic Digestion of Vegetable Crop Residues. *Energies* **2019**, *12*, (1), 26.
- 85. Yan, H.; Zhao, C.; Zhang, J.; Zhang, R.; Xue, C.; Liu, G.; Chen, C., Study on biomethane
- production and biodegradability of different leafy vegetables in anaerobic digestion. *AMB Express* **2017**, 852 7, (1), 27.
- 853 86. Lakhundi, S.; Siddiqui, R.; Khan, N. A., Cellulose degradation: a therapeutic strategy in the
- improved treatment of Acanthamoeba infections. *Parasites & Vectors* **2015**, *8*, (1), 23.
- 855 87. Note: dodecyl ester CNF and DA-CNF-2 are the same sample used in two different data sets.
- 856 88. Homma, I.; Fukuzumi, H.; Saito, T.; Isogai, A., Effects of carboxyl-group counter-ions on
- biodegradation behaviors of TEMPO-oxidized cellulose fibers and nanofibril films. *Cellulose* **2013**, *20*, (5), 2505-2515.
- 859 89. Homma, I.; Isogai, T.; Saito, T.; Isogai, A., Degradation of TEMPO-oxidized cellulose fibers and nanofibrils by crude cellulase. *Cellulose* **2013**, *20*, (2), 795-805.
- Haske-Cornelius, O.; Pellis, A.; Tegl, G.; Wurz, S.; Saake, B.; Ludwig, R.; Sebastian, A.;
- Nyanhongo, G. S.; Guebitz, G. M., Enzymatic Systems for Cellulose Acetate Degradation. *Catalysts* **2017**, *7*, (10), 287.
- 2017, 7, (10), 287.
- White, G. F.; Russell, N. J.; Tidswell, E. C., Bacterial scission of ether bonds. *Microbiol Rev*
- **1996,** *60*, (1), 216-232.
- 866 92. Teeri, T. T., Crystalline cellulose degradation: new insight into the function of
- cellobiohydrolases. *Trends in Biotechnology* **1997**, *15*, (5), 160-167.
- 93. Johansson, L.-S.; Tammelin, T.; Campbell, J. M.; Setala, H.; Osterberg, M., Experimental
- evidence on medium driven cellulose surface adaptation demonstrated using nanofibrillated cellulose. *Soft Matter* **2011,** *7*, (22), 10917-10924.
- 871 94. Cumpstey, I., Chemical modification of polysaccharides. ISRN Org Chem 2013, 2013, 417672-
- 872 417672.
- 873 95. Heinze, T.; El Seoud, O. A.; Koschella, A., Cellulose Derivatives: Synthesis, Structure, and
- 874 *Properties.* Springer International Publishing: Switzerland, 2018.
- 875 96. Huang, F.-Y., Thermal Properties and Thermal Degradation of Cellulose Tri-Stearate (CTs).
- 876 *Polymers* **2012**, *4*, (2), 1012-1024.
- 877 97. Vaca-Garcia, C.; Thiebaud, S.; Borredon, M. E.; Gozzelino, G., Cellulose esterification with fatty
- acids and acetic anhydride in lithium chloride/N,N-dimethylacetamide medium. *Journal of the American*
- 879 *Oil Chemists' Society* **1998,** 75, (2), 315-319.
- 880 98. Okazaki, M.; Moo-Young, M., Kinetics of enzymatic hydrolysis of cellulose: Analytical
- description of a mechanistic model. *Biotechnology and Bioengineering* **1978**, *20*, (5), 637-663.
- 882 99. Gaikwad, A., Effect of Particle Size on the Kinetics of Enzymatic Hydrolysis of Microcrystalline
- Cotton Cellulose: a Modeling and Simulation Study. *Applied Biochemistry and Biotechnology* **2019**, *187*,
- 884 (3), 800-816.
- 885 100. Vickerman, J. C.; GIlmore, I. S., Surface Analysis: The Principal Techniques. Second ed.; Wiley:
- 886 2009.
- Petersen, E. J.; Zhang, L.; Mattison, N. T.; O'Carroll, D. M.; Whelton, A. J.; Uddin, N.; Nguyen,
- T.; Huang, Q.; Henry, T. B.; Holbrook, R. D.; Chen, K. L., Potential Release Pathways, Environmental
- Fate, And Ecological Risks of Carbon Nanotubes. *Environ. Sci. Technol.* **2011**, *45*, (23), 9837-9856.
- 890 102. Bai, Y.; Wu, F.; Lin, D.; Xing, B., Aqueous stabilization of carbon nanotubes: effects of surface
- 891 oxidization and solution chemistry. Environmental Science and Pollution Research 2014, 21, (6), 4358-
- 892 4365.

- 893 103. Frank, B. P.; Sigmon, L. R.; Deline, A. R.; Lankone, R. S.; Gallagher, M. J.; Zhi, B.; Haynes, C.
- 894 L.; Fairbrother, D. H., Photochemical Transformations of Carbon Dots in Aqueous Environments.
- 895 Environ. Sci. Technol. **2020**, *54*, (7), 4160-4170.
- 896 104. Zhang, B.; Liu, C.-y.; Liu, Y., A Novel One-Step Approach to Synthesize Fluorescent Carbon
- Nanoparticles. European Journal of Inorganic Chemistry 2010, 2010, (28), 4411-4414.
- 898 105. Zhao, P.; Zhu, L., Dispersibility of carbon dots in aqueous and/or organic solvents. *Chemical*
- 899 *Communications* **2018,** *54*, (43), 5401-5406.
- 900 106. Nuncira, J.; Seara, L. M.; Sinisterra, R. D.; Caliman, V.; Silva, G. G., Long-term colloidal
- stability of graphene oxide aqueous nanofluids. *Fullerenes, Nanotubes and Carbon Nanostructures* 2019,
   1-11.
- 903 107. Qi, Y.; Xia, T.; Li, Y.; Duan, L.; Chen, W., Colloidal stability of reduced graphene oxide
- materials prepared using different reducing agents. *Environ. Sci.: Nano* **2016**, *3*, (5), 1062-1071.
- 905 108. Baker, A. A.; Helbert, W.; Sugiyama, J.; Miles, M. J., New Insight into Cellulose Structure by
- Atomic Force Microscopy Shows the Iα Crystal Phase at Near-Atomic Resolution. *Biophys J* 2000, 79,
  (2), 1139-1145.
- 908 109. Stern, J.; Moraïs, S.; Lamed, R.; Bayer, E. A., Adaptor Scaffoldins: An Original Strategy for
- 909 Extended Designer Cellulosomes, Inspired from Nature. *mBio* **2016**, 7, (2), e00083-16.
- 910 110. Wang, Y.; Leng, L.; Islam, M. K.; Liu, F.; Lin, C. S. K.; Leu, S.-Y., Substrate-Related Factors
- Affecting Cellulosome-Induced Hydrolysis for Lignocellulose Valorization. *Int J Mol Sci* 2019, 20, (13),
   3354.
- 913 111. Aulin, C.; Ahola, S.; Josefsson, P.; Nishino, T.; Hirose, Y.; Österberg, M.; Wågberg, L.,
- 914 Nanoscale Cellulose Films with Different Crystallinities and Mesostructures—Their Surface Properties
- and Interaction with Water. *Langmuir* **2009**, *25*, (13), 7675-7685.
- 916 112. Ahola, S.; Salmi, J.; Johansson, L. S.; Laine, J.; Österberg, M., Model Films from Native
- 917 Cellulose Nanofibrils. Preparation, Swelling, and Surface Interactions. *Biomacromolecules* **2008**, *9*, (4),
- 918 1273-1282.
- 919 113. Spence, K. L.; Venditti, R. A.; Rojas, O. J.; Habibi, Y.; Pawlak, J. J., The effect of chemical
- omposition on microfibrillar cellulose films from wood pulps: water interactions and physical properties
- 921 for packaging applications. *Cellulose* **2010**, *17*, (4), 835-848.
- 922 114. Westerholm, M.; Schnurer, A., Microbial Responses to Different Operating Practices for Biogas
- Production Systems. In *Anaerobic Digestion*, Banu, R., Ed. IntechOpen: 2019.
- 924 115. Mahamat, A. Y.; Gourdon, R.; Leger, P.; Vermande, P., Methane recovery by anaerobic digestion
- 925 of cellulosic materials available in Sahel. *Biological Wastes* **1989**, *30*, (3), 181-197.
- 926 116. Komarek, R. J.; Gardner, R. M.; Buchanan, C. M.; Gedon, S., Biodegradation of radiolabeled
- 927 cellulose acetate and cellulose propionate. J. Appl. Polym. Sci. 1993, 50, (10), 1739-1746.
- 928 117. Wei, L.; Agarwal, U. P.; Hirth, K. C.; Matuana, L. M.; Sabo, R. C.; Stark, N. M., Chemical
- 929 modification of nanocellulose with canola oil fatty acid methyl ester. *Carbohydr. Polym.* **2017**, *169*, 108-930 116.
- 931 118. Ferreira, F. V.; Pinheiro, I. F.; Gouveia, R. F.; Thim, G. P.; Lona, L. M. F., Functionalized
- 932 cellulose nanocrystals as reinforcement in biodegradable polymer nanocomposites. Polym. Compos. 2018,
- 933 *39*, (S1), E9-E29.
- 934 119. Kargarzadeh, H.; Huang, J.; Lin, N.; Ahmad, I.; Mariano, M.; Dufresne, A.; Thomas, S.; Gałeski,
- 935 A., Recent developments in nanocellulose-based biodegradable polymers, thermoplastic polymers, and
- porous nanocomposites. *Progress in Polymer Science* **2018**, 87, 197-227.
- 937 120. Mishra, R. K.; Sabu, A.; Tiwari, S. K., Materials chemistry and the futurist eco-friendly
- applications of nanocellulose: Status and prospect. *Journal of Saudi Chemical Society* 2018, 22, (8), 949-939
   978.
- 940 121. Rodionova, G.; Lenes, M.; Eriksen, Ø.; Gregersen, Ø., Surface chemical modification of
- microfibrillated cellulose: improvement of barrier properties for packaging applications. Cellulose 2011,
- 942 18, (1), 127-134.

- 943 122. Yagyu, H.; Ifuku, S.; Nogi, M., Acetylation of optically transparent cellulose nanopaper for high
- thermal and moisture resistance in a flexible device substrate. Flexible and Printed Electronics 2017, 2,
- 945 (1), 014003.

- 946 123. Yang, S.; Xie, Q.; Liu, X.; Wu, M.; Wang, S.; Song, X., Acetylation improves thermal stability
- and transmittance in FOLED substrates based on nanocellulose films. RSC Adv. 2018, 8, (7), 3619-3625.
- 948 124. Barhoum, A.; Samyn, P.; Öhlund, T.; Dufresne, A., Review of recent research on flexible
- 949 multifunctional nanopapers. *Nanoscale* **2017**, *9*, (40), 15181-15205.
- 950 125. Mertaniemi, H.; Laukkanen, A.; Teirfolk, J.-E.; Ikkala, O.; Ras, R. H. A., Functionalized porous
- 951 microparticles of nanofibrillated cellulose for biomimetic hierarchically structured superhydrophobic
- 952 surfaces. RSC Adv. 2012, 2, (7), 2882-2886.
- 953 126. Zhang, Y.; Wei, L.; Hu, H.; Zhao, Z.; Huang, Z.; Huang, A.; Shen, F.; Liang, J.; Qin, Y.,
- Tribological properties of nano cellulose fatty acid esters as ecofriendly and effective lubricant additives.
- 955 *Cellulose* **2018**, *25*, (5), 3091-3103.

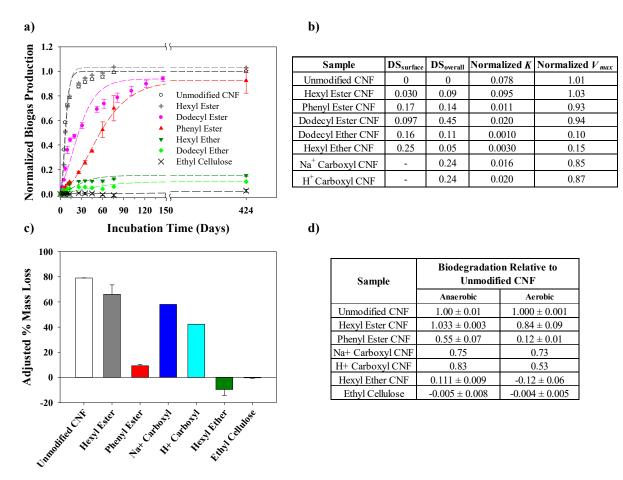
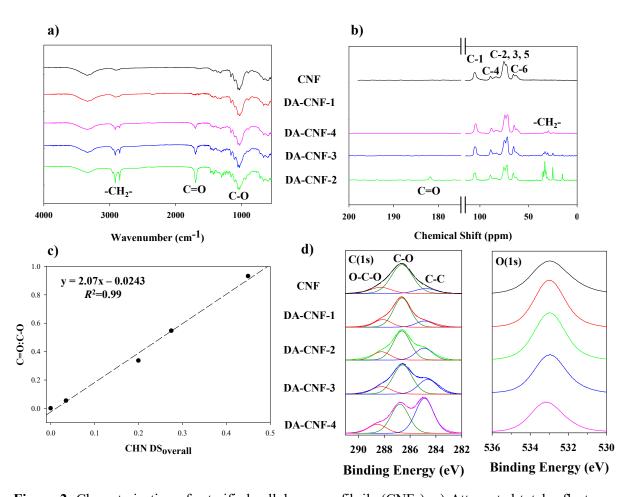
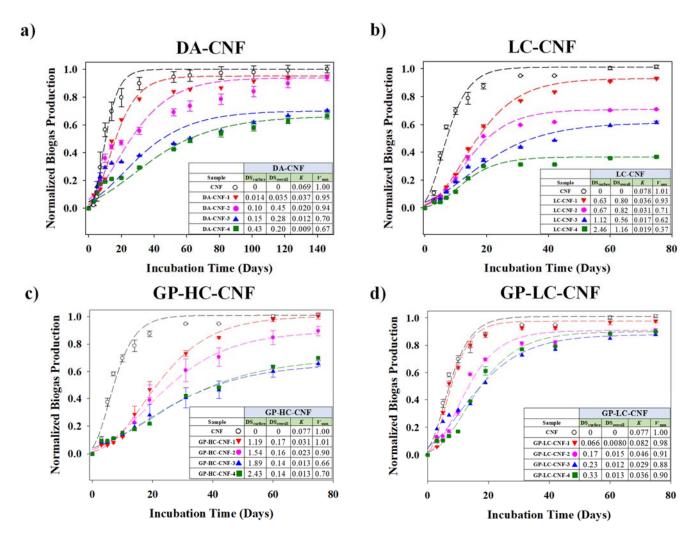


Figure 1. Anaerobic and aerobic biodegradation of functionalized cellulose nanofibrils (CNF). a) Normalized biogas production by an anaerobic microbial community degrading the indicated CNFs as a function of incubation time (dodecyl ester CNF was not sampled past 146 days). For each sample, values are normalized to the maximum calculated biogas produced by both cellulose and the added functional group (see text for details). Error bars represent one standard deviation from duplicate samples. b)  $DS_{surface}$  and  $DS_{overall}$  values determined by XPS and elemental analysis, respectively and K and  $V_{max}$  derived from modified Gompertz model fitting of biodegradation data for functionalized CNFs. c) Mass loss at 60 d for degradation of the indicated CNFs by an aerobic microbial community. Values shown represent the difference between microbial and blank samples (see experimental section for details). Error bars represent one standard deviation from duplicate samples. Na<sup>+</sup> and H<sup>+</sup> Carboxyl CNF samples were not run in duplicate due to limitations in sample mass. d) Comparison of the inhibition of biodegradation for different functionalized CNFs in anaerobic vs. aerobic environments relative to unmodified CNF.



**Figure 2.** Characterization of esterified cellulose nanofibrils (CNFs). a) Attenuated total reflectance Fourier transform infrared spectroscopy (ATR-FTIR) of unmodified CNF (black), and DA-CNF-1 (red), DA-CNF-4 (pink), DA-CNF-3 (blue), and DA-CNF-2 (green) dodecyl ester CNFs. The unmodified CNF contains the characteristic O–H (3339 cm<sup>-1</sup>), C–H (2905 cm<sup>-1</sup>), and C–O (1031 cm<sup>-1</sup>) stretches of cellulose with CH<sub>2</sub> and C=O peaks due to the ester linkages b) Solid-state <sup>13</sup>C-NMR spectra of unmodified CNFs (black), DA-CNF-4 (pink), DA-CNF-3 (blue), and DA-CNF-2 (green) dodecyl ester CNFs. c) Relation between vibrational peak ratio (C=O:C–O) and overall DS calculated from CHN elemental analysis %C data. d) X-ray photoelectron C(1s) and O(1s) spectra of unmodified CNFs (black), DA-CNF-1 (red), DA-CNF-2 (green), DA-CNF-3 (blue), and DA-CNF-4 (pink) dodecyl ester CNFs.



**Figure 3.** Biogas production and modified Gompertz model fits (dotted lines) of cellulose nanofibrils (CNFs) esterified using liquid- and gas-phase methods. Relative biogas production from cellulose nanofibrils esterified using a) liquid-phase dodecanoic acid (DA-CNF), b) liquid-phase lauroyl chloride (LC-CNF), c) gas-phase hexanoyl chloride (GP-HC-CNF), and d) gas-phase lauroyl chloride (GP-LC-CNF). Biogas volumes are normalized to the maximum calculated biogas production expected from the combined cellulose and functional group components of each sample (Table S4). The  $DS_{surface}$  and  $DS_{overall}$  values, as well as normalized maximum biogas volume ( $V_{max}$ ) and biogas production rate (K) for each sample are provided in the inset of each plot. Accuracy for Gompertz parameters is reported in Table S5.