

Tl₂Ir₂O₇: A Pauli Paramagnetic Metal, Proximal to a Metal Insulator Transition

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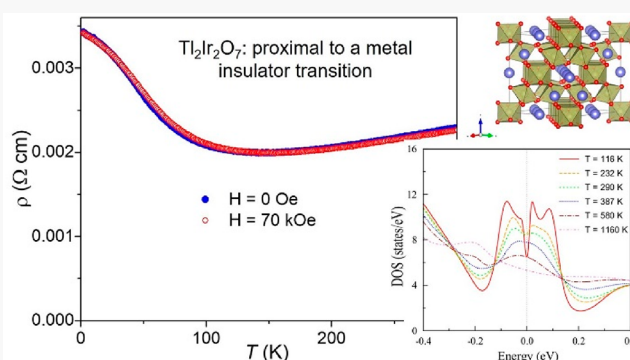
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ABSTRACT: A polycrystalline sample of Tl₂Ir₂O₇ was synthesized by high-pressure and high-temperature methods. Tl₂Ir₂O₇ crystallizes in the cubic pyrochlore structure with space group *Fd3m* (No. 227). The Ir⁴⁺ oxidation state is confirmed by Ir–L₃ X-ray absorption near-edge spectroscopy. Combined temperature-dependent magnetic susceptibility, resistivity, specific heat, and DFT+DMFT calculation data show that Tl₂Ir₂O₇ is a Pauli paramagnetic metal, but it is close to a metal–insulator transition. The effective ionic size of Tl³⁺ is much smaller than that of Pr³⁺ in metallic Pr₂Ir₂O₇; hence, Tl₂Ir₂O₇ would be expected to be insulating according to the established phase diagram of the pyrochlore iridate compounds, A³⁺₂Ir₂O₇. Our experimental and theoretical studies indicate that Tl₂Ir₂O₇ is uniquely different from the current A³⁺₂Ir₂O₇ phase diagram. This uniqueness is attributed primarily to the electronic configuration difference between Tl³⁺ and rare-earth ions, which plays a substantial role in determining the Ir–O–Ir bond angle, and the corresponding electrical and magnetic properties.



INTRODUCTION

The 5d oxides are attracting intensive studies owing to their fascinating physical properties, such as a Mott insulating state driven by spin–orbit coupling (SOC) in Sr₂IrO₄,¹ a ferroelectric metal LiOsO₃,² metal–insulator transition (MIT),³ superconductor,⁴ and high-temperature ferromagnetic or ferrimagnetic insulator/half-metal materials.^{5–8} Compared to 3d electrons, 5d electrons present much stronger SOC, weaker Coulomb interactions, and thus provide a unique platform to study correlated electronic systems.⁹

The iridium pyrochlores, A³⁺₂Ir₂O₇ (A = Bi, Y, or a lanthanide ion), display a fascinating phase diagram.^{10–31} Studies of these iridium pyrochlores indicate that the physical properties are generally correlated to the ionic size of A³⁺ ions, which determine the trigonal distortions of IrO₆ octahedra and Ir–O–Ir bond angles, thereby affecting the hopping between Ir electrons.^{9,32} Bi₂Ir₂O₇ and Pr₂Ir₂O₇ with relatively large A³⁺ cations are metallic,^{10,16,19,25} while others with smaller A³⁺ ions are antiferromagnetically ordered and electrically insulating at low temperatures,^{17,18,21,27,31} particularly, Nd₂Ir₂O₇, Sm₂Ir₂O₇, and Eu₂Ir₂O₇ show MIT.^{12,21,28–30}

Here, we report the synthesis, crystal structure, and magnetic and electrical properties of a new iridate pyrochlore, Tl₂Ir₂O₇. The Shannon effective ionic radius of Tl³⁺ (0.98 Å) is comparable to those of Yb³⁺ (0.985 Å) and Lu³⁺ (0.977 Å),

and an insulating state is expected for Tl₂Ir₂O₇, as both Yb₂Ir₂O₇ and Lu₂Ir₂O₇ are insulating. However, our studies find that Tl₂Ir₂O₇ is a Pauli paramagnetic metal and is close to a MIT.³³

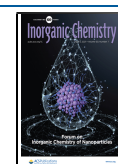
EXPERIMENTAL SECTION

Sample Preparation. Powders of Tl₂O₃ (Alfa, 99.9%) and IrO₂ (Alfa, 99.99%) with a molar ratio of 1:2 were weighed and mixed well using a mortar and pestle. The IrO₂ was heated overnight at 1173 K before using. The powders were then sealed in a Pt capsule, which was placed inside a MgO crucible. They were statically compressed at a pressure of 5 GPa in a Walker-type multianvil press and heated at 1373 K for 2 h before quenching to ambient pressure while keeping the pressure stable. Then the pressure was released gradually.

Synchrotron Powder X-ray Diffraction (SPXD). Part of the Tl₂Ir₂O₇ sample was ground to a fine powder, which was then characterized by SPXD (λ = 0.41273 Å) at ambient temperature at Beamline 11-BM of the Advanced Photon Source (APS) of Argonne National Laboratory. Rietveld refinements of the powder diffraction

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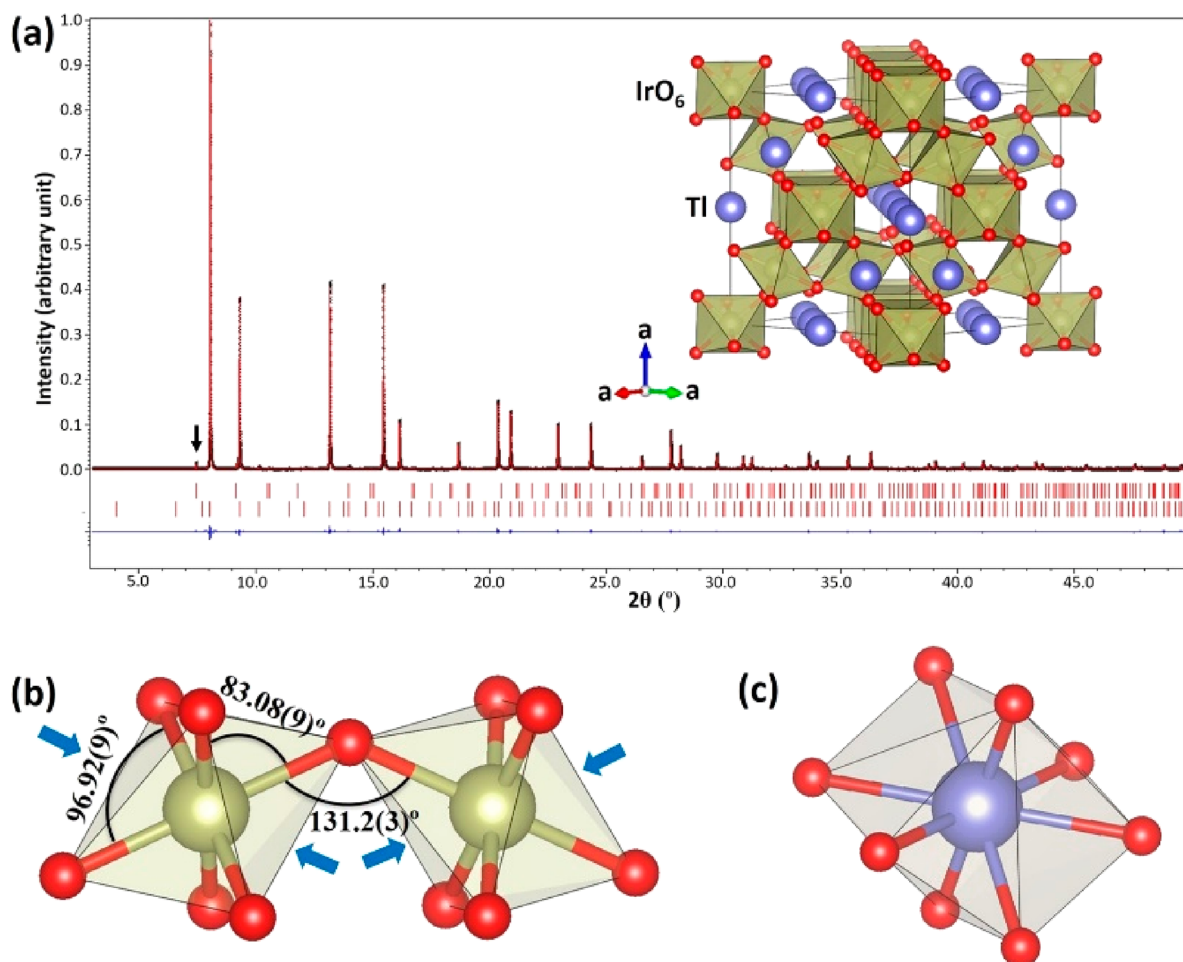


Figure 1. (a) Rietveld refined room-temperature SPXD pattern of $\text{Tl}_2\text{Ir}_2\text{O}_7$. The black asterisks are experimentally observed data. The calculated patterns are shown in red, and the blue patterns at the bottom are their difference. The lower vertical red bars indicate the diffraction positions for $\text{Tl}_2\text{Ir}_2\text{O}_7$ and the upper ones for the minor impurity IrO_2 (1.0% volume fraction). The strongest IrO_2 diffraction peak is marked with a black arrow. The crystal structure of $\text{Tl}_2\text{Ir}_2\text{O}_7$ is shown as an inset. (b) IrO_6 octahedra. The blue arrows indicate the trigonal compression of IrO_6 octahedra. (c) TlO_8 coordination in $\text{Tl}_2\text{Ir}_2\text{O}_7$.

data were carried out with the JANA2006 software,³⁴ and the crystal structures were drawn with the VESTA software.³⁵

X-ray Absorption Near-Edge Spectroscopy (XANES). The XANES measurements were performed at the national synchrotron light source (NSLS-I and NSLS-II) at Brookhaven National Laboratory. The standard spectra were mostly collected at NSLS-I on beamlines X19A (X18B) with a Si-111 double-crystal (channel cut) monochromator. The measurement on $\text{Tl}_2\text{Ir}_2\text{O}_7$ was performed at the next-generation NSLS-II facility on the 6-BM, BMM beamline with a Si-311 double-crystal monochromator. In the data collection, both transmission mode and fluorescent measurements were made with simultaneous standards for precision energy calibration. Standard linear background and postedge normalization to unity absorption step height were used in the analysis.^{36–46}

Magnetism and Electrical Resistivity. The temperature dependence of the magnetic susceptibility was measured in a VSM-7T (Quantum Design). The measurements were conducted under zero-field cooling (ZFC) and field-cooling (FC) conditions, respectively, in the temperature range of 2–300 K under an applied magnetic field of 1 and 70 kOe. Temperature-dependent electrical resistivity was measured with a direct current (dc) gauge current of 0.1 mA by the four-point method using a PPMS-14 (Quantum Design).

Density Functional Theory Plus Dynamical Mean-Field Theory (DFT+DMFT). Fully charge self-consistent DFT+DMFT calculations^{47–49} implemented in Wien2k package⁵⁰ were performed with formalisms described in ref 51. The experimental crystal

structure of $\text{Tl}_2\text{Ir}_2\text{O}_7$ was adopted for the calculations. In DFT calculations, the generalized gradient approximation (GGA) of Perdew–Burke–Ernzerhof (PBE) was used for the exchange–correlation functional. The muffin tin radii were chosen to be 2.28, 2.04, and 1.67 Bohr radii for Tl, Ir, and O, respectively; the size of a plane-wave basis set was determined from $R_{\text{mt}}K_{\text{max}}$ of 7.0, where R_{mt} is the smallest atomic muffin tin radius and K_{max} is the largest plane-wave vector. A $17 \times 17 \times 17$ k -point mesh was used for the Brillouin zone integration. In DMFT calculations, the continuous-time quantum Monte Carlo method (CTQMC)^{52,53} was implemented for a local impurity solver with a proper local J effective basis minimizing the sign problem. We choose a wide hybridization energy window from -10 to 10 eV with respect to the Fermi level E_F . The fully rotational invariant form was applied for a local Coulomb interaction Hamiltonian with on-site Coulomb repulsion $U = 4.5$ eV and $J_H = 0.8$ eV, where the U and J_H values were adopted in previous DFT+DMFT calculations on pyrochlore iridate compounds.⁵⁴ The maximum entropy method⁵⁵ was used for analytical continuation to obtain the self-energy on the real frequency.

RESULTS

Crystal Structure. The SPXD data of $\text{Tl}_2\text{Ir}_2\text{O}_7$ collected at room temperature (see Figure 1a) can be refined well with a cubic pyrochlore crystal structure, space group $Fd\bar{3}m$ (No. 227), which was reported for other iridium pyrochlores.²² There was about 1.0% volume fraction of IrO_2 impurity found

Table 1. Refined Atomic Positions and Displacement Parameters of $\text{Ti}_2\text{Ir}_2\text{O}_7$ at Room Temperature^a

atom	Wyckoff positions	occupancy	x	y	z	U_{iso} ($\text{\AA}^2$)
Tl	16d	1	0.5	0.5	0.5	0.0048(2)
Ir	16c	1	0	0	0	0.0016(2)
O1	48f	1	0.3307(4)	0.125	0.125	0.0069(9)
O2	8b	1	0.375	0.375	0.375	0.013(3)

^aSpace group $Fd\bar{3}m$ (No. 227), $a = 10.18258(5)$ $\text{\AA}$, $Z = 8$, $V = 1055.779(9)$ $\text{\AA}^3$, and $d = 11.392$ g/cm^3 . $R_{\text{wp}} = 3.74\%$, and goodness of fit is 1.99.

in the sample. The cell parameter $a = 10.18258(5)$ $\text{\AA}$ is comparable to that of 10.184–10.186 $\text{\AA}$ reported for $\text{Ti}_2\text{Ru}_2\text{O}_7$ synthesized at 1–3 GPa.⁵⁶ In $\text{Ti}_2\text{Ir}_2\text{O}_7$, Tl ions occupy the 16d ($1/2, 1/2, 1/2$) sites, Ir ions occupy the 16c (0, 0, 0) sites, and O anions occupy 48f ($x, 1/8, 1/8$) and 8b ($3/8, 3/8, 3/8$) sites. During the refinement, there was no indication of Tl/Ir antisite exchange, and there is no indication for a vacancy. We thus constrained the occupancies to 1 for all the atoms in the last refinement. The final refined atomic positions and displacement parameters are shown in Table 1. In $\text{Ti}_2\text{Ir}_2\text{O}_7$, the IrO_6 octahedra are trigonally distorted as shown in Figure 1b, and the Ir–O–Ir bond angle is 131.2°. Tl ions are coordinated with eight O forming TlO_8 polyhedra (see Figure 1c).

The bond lengths of Tl–O and Ir–O within the TlO_8 polyhedra and IrO_6 octahedra are summarized in Table 2. The

Table 2. Selected Bond Length for $\text{Ti}_2\text{Ir}_2\text{O}_7$

bond	bond length ($\text{\AA}$)
Ir–O1	$1.979(3) \times 6$
Tl–O1	$2.492(3) \times 6$
Tl–O2	2.2046×2

average Tl–O bond length, 2.420 $\text{\AA}$, is similar to values for $\text{Ti}_2\text{Mn}_2\text{O}_7$ (2.38 $\text{\AA}$),^{57–59} for TiNiO_3 (2.42 $\text{\AA}$),^{60,61} for TiCrO_3 (2.43 $\text{\AA}$),^{62,63} for TiMnO_3 (2.46 $\text{\AA}$),⁶⁴ and for $\text{Ti}_2\text{Ru}_2\text{O}_7$ ^{56,65} (2.49 $\text{\AA}$) with Ti^{3+}O_8 coordination, supporting the nominal Ti^{3+} valence state. The average Ir–O bond distances 1.979 $\text{\AA}$, which is in reasonable agreement with those for $\text{Y}_2\text{Ir}_2\text{O}_7$ ²⁷ (1.98 $\text{\AA}$) and for $\text{Pr}_2\text{Ir}_2\text{O}_7$ ²² (2.01 $\text{\AA}$), and $\text{Nd}_2\text{Ir}_2\text{O}_7$ ¹³ with Ir^{4+}O_6 coordination. To further confirm the valence state of Ir in $\text{Ti}_2\text{Ir}_2\text{O}_7$ XANES was investigated (*vide infra*).

Ir– L_3 XANES. A comprehensive comparison of the Ir– L_3 “white line” (WL) feature of $\text{Ti}_2\text{Ir}_2\text{O}_7$ to those of the standard spectra for $\text{Sr}_2\text{CaIrO}_6$ (Ir^{6+}), $\text{Sr}_2\text{ScIrO}_6$ (Ir^{5+}), Sr_2YIrO_6 (Ir^{5+}), IrO_2 (Ir^{4+}), and $\text{La}_2\text{CuIrO}_6$ (Ir^{4+}) is shown in Figure 2a. An expanded and simultaneous standard-based calibrated energy scale has been used. The WL feature shows a consistent chemical shift toward lower energy from $\sim\text{Ir}^{6+}$ to $\sim\text{Ir}^{5+}$ to $\sim\text{Ir}^{4+}$ standard spectra, and the chemical shift of the $\text{Ti}_2\text{Ir}_2\text{O}_7$ spectrum is consistent with the energy for the $\sim\text{Ir}^{4+}$ standard spectrum.

For 5d oxides, the spectral structure of these WL features can also be employed as valence/configuration indicators. The WL spectral structure of octahedrally-O transition metals (TM)– L_3 involves a prominent bimodal A/B semiresolved peak. For a large class of 5d TM oxides, this structure originates from the splitting of d orbitals into lower 6-fold degenerate t_{2g} and higher 4-fold degenerate e_g multiplets due to the octahedral–O coordination.^{36–46} Specifically, the bimodal A/B features are correlated to transitions into t_{2g}/e_g final hole states, respectively. The A(t_{2g} -hole-related)/B(e_g -hole-related) features are best demonstrated in a $5d^0$ $\text{Ca}_2\text{MnTaO}_6$ standard spectrum³⁸ in Figure 2b. The dramatic

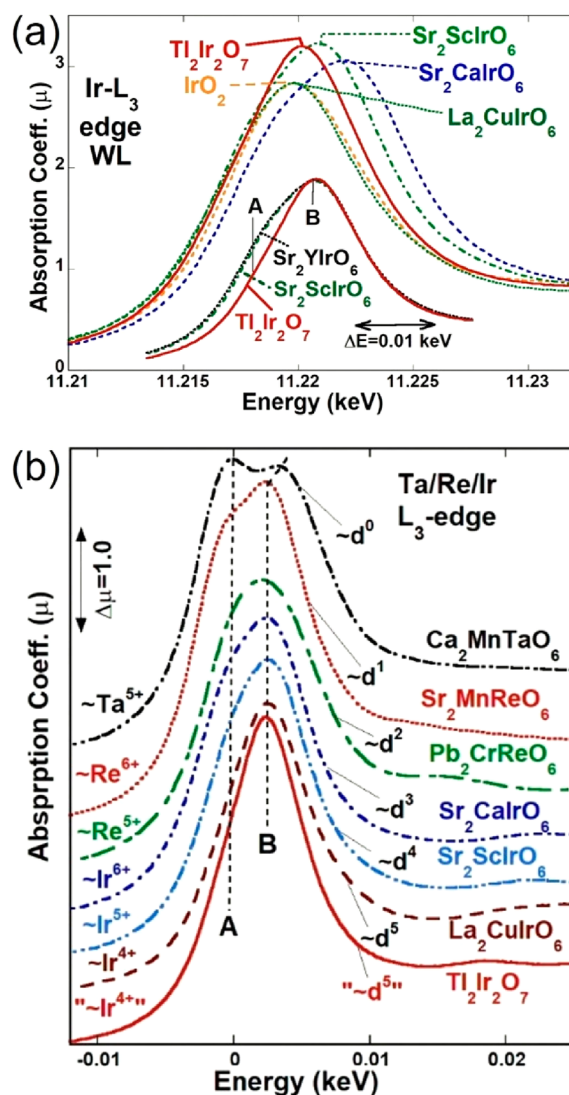


Figure 2. (a) Ir– L_3 WL feature of $\text{Ti}_2\text{Ir}_2\text{O}_7$ compared with the spectra of $\text{Sr}_2\text{CaIrO}_6$ (Ir^{6+}), $\text{Sr}_2\text{ScIrO}_6$ (Ir^{5+}), Sr_2YIrO_6 (Ir^{5+}), IrO_2 (Ir^{4+}), and $\text{La}_2\text{CuIrO}_6$ (Ir^{4+}). Inset shows an enlarged spectrum of $\text{Ti}_2\text{Ir}_2\text{O}_7$ superimposed with $\sim\text{Ir}^{5+}$ standard spectra to demonstrate the decrease of A-feature intensity for $\text{Ti}_2\text{Ir}_2\text{O}_7$. It should be noted that the standard spectra have been shifted to compensate for the chemical shift and rescaled for superposition. (b) Ir– L_3 WL-feature is compared with a series of standard oxides with $5d^0$ to $5d^5$ configurations. The bimodal A/B structure is correlated to transitions into t_{2g}/e_g final states, respectively; and the systematic relative loss in A-feature intensity with decreasing t_{2g} state holes.

and systematic loss of the intensity of the A-feature, with increasing t_{2g} orbital electrons, is clearly shown in Figure 2b for a series of standard spectra spanning $5d^0$ – $5d^5$.^{36–46} The relative A-feature intensity for the $\text{Ti}_2\text{Ir}_2\text{O}_7$ Ir– L_3 spectrum is extremely small and comparable to that of the d^5 standard

$\text{La}_2\text{CuIrO}_6$.^{44,45} For clarification, the spectrum of $\text{Tl}_2\text{Ir}_2\text{O}_7$ is shown on an expanded scale in the inset of Figure 2a superimposed with the spectra of $\sim\text{Ir}^{5+}$ (d^4) standards, where the standard spectra were shifted to compensate for the chemical shift and rescaled for superposition. In comparison to the $\sim\text{Ir}^{5+}$ standards, the smaller A-feature intensity of the $\text{Tl}_2\text{Ir}_2\text{O}_7$ spectrum (compared to the $\sim\text{Ir}^{5+}$ standards) is demonstrated. The $\text{Tl}_2\text{Ir}_2\text{O}_7$ spectrum is sharper since it was measured on the NSLS-II beamline 6-BM, which implemented a Si-311 monochromator for higher resolution. Thus, the comparisons of the XANES spectra support a $\sim\text{Ir}^{4+}$ configuration for $\text{Tl}_2\text{Ir}_2\text{O}_7$ from the chemical shift as well as WL A/B feature perspectives.

Magnetic Properties. The temperature-dependent magnetic susceptibility curves, $\chi(T)$, are shown in Figure 3a. The

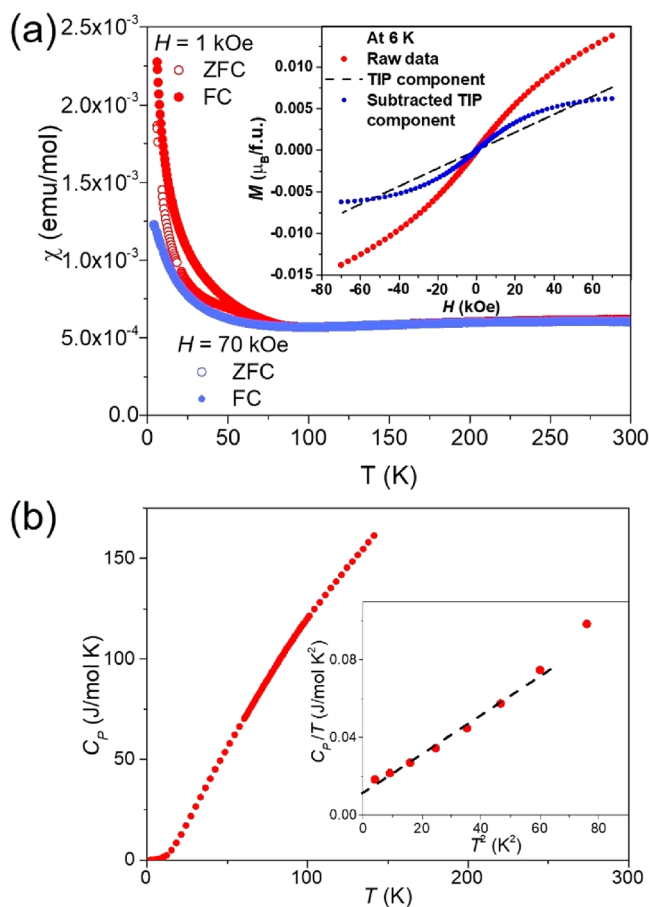


Figure 3. (a) Temperature-dependent magnetic susceptibility curves of $\text{Tl}_2\text{Ir}_2\text{O}_7$ measured at 1 and 70 kOe. The inset exhibits the field-dependent magnetization curves of $\text{Tl}_2\text{Ir}_2\text{O}_7$ measured at 6 K, and the data corrected for a temperature-independent (TIP) component. (b) Temperature-dependent specific heat curve of $\text{Tl}_2\text{Ir}_2\text{O}_7$. The inset exhibits the corresponding C_p/T vs T^2 data below 9 K.

$\chi(T)$ curves measured at 1 and 70 kOe are temperature-independent from 300–75 K, indicating a Pauli paramagnetism with a temperature-independent term $\chi_{\text{TIP}} \approx 5.7 \times 10^{-4}$ emu/mol (see the dashed lines in Figure 3a). This χ_{TIP} value is comparable to a Pauli paramagnetic value, $\chi_p \approx 5.0 \times 10^{-4}$ emu/mol, reported in other Ir pyrochlores $\text{Pr}_2\text{Ir}_2\text{O}_7$ ²³ and $(\text{Y}, \text{Ca})_2\text{Ir}_2\text{O}_7$.⁶⁶ The upturn of the $\chi(T)$ curves below 75 K is possibly due to magnetic impurity. The divergence of ZFC and FC curves measured at 1 kOe was suppressed when measured

at 70 kOe, which indicates that the magnetic impurity contribution can, as expected, be saturated in a magnetic field. Analysis of the 6 K field-dependent magnetization curves, $M(H)$, also supports that the upturn of the $\chi(T)$ curves is due to a small amount of ferromagnetic impurity as the magnetic impurity contribution is almost saturated at 70 kOe with a tiny magnetization value of $0.006 \mu_B/\text{f.u.}$ as shown in the inset of Figure 3a. Specific heat was then measured from 140 to 2 K (see Figure 3b), and the specific heat curve does not show any noticeable anomaly, indicating that the upturn of the $\chi(T)$ curve is not related to magnetic order. Therefore, the low-temperature upturn of $\chi(T)$ may be related to the magnetic impurities. It should be noted that (as in the case of the Pauli paramagnetic perovskite BaIrO_3)⁶⁷ in the temperature range above where the Curie tail dominates that there is a perceptible minimum in $\chi(T)$ near 100 K.

Electrical Resistivity. Figure 4 shows the temperature-dependent electrical resistivity $\rho(T)$ curves. The ρ is about 2.3

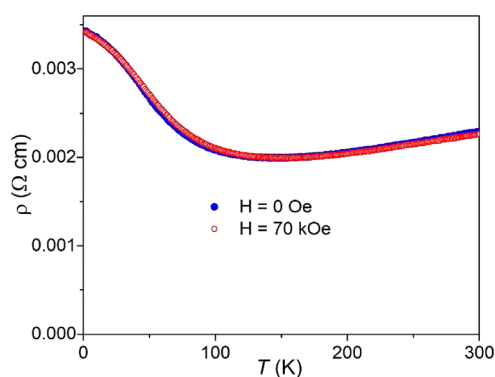


Figure 4. Temperature-dependent electrical resistivity of $\text{Tl}_2\text{Ir}_2\text{O}_7$.

mΩ cm at room temperature, which is of the same magnitude as that of $\text{Pr}_2\text{Ir}_2\text{O}_7$. $\rho(T)$ shows metallic behavior down to 150 K with the ρ decreasing with cooling, then the ρ gradually increases down to 2 K, suggesting a possible MIT. The C_p/T versus T^2 data at low temperatures (see the inset of Figure 3b) show approximately linear behavior and was fitted with the approximate Debye model $C = \gamma T + \beta T^3$, where the γ is the electronic specific heat coefficient and β is a constant related to the Debye temperature Θ_D ($\sim \beta^{-1/3}$). The fitting resulted in an estimated γ of 12(1) mJ mol^{−1} K^{−2}. The nonzero γ of 12(1) mJ mol^{−1} K^{−2}, comparable to or larger than that of 6.8–9.3 mJ mol^{−1} K^{−2} for metallic BaIrO_3 ⁶⁷ and AOsO_3 ($A = \text{Ca}, \text{Sr}, \text{and Ba}$),⁶⁸ indicates that a charge gap may not be open in $\text{Tl}_2\text{Ir}_2\text{O}_7$. Thus, the upturn of ρ at low temperatures may be related to the fact that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is close to a MIT, which is further supported by the DFT+DMFT calculations showing strong electronic correlations in $\text{Tl}_2\text{Ir}_2\text{O}_7$ (*vide infra*).

DFT+DMFT Calculations. The (dimensionless) Wilson ratio, R_W , is useful to identify how strong the electronic correlation in $\text{Tl}_2\text{Ir}_2\text{O}_7$ is, and it is estimated from the measured magnetic susceptibility and electronic specific heat. From the experimental data of $\chi = 5.7 \times 10^{-4}$ emu/mol and $\gamma = 12$ mJ mol^{−1} K^{−2}, $R_W = 3.6$ is obtained, which is significantly away from 1 indicating the strong electron–electron interactions. Comparable R_W values have been reported for $\text{CaCu}_3\text{Ru}_4\text{O}_{12}$ (2.9)⁶⁹ and for CeCu_6 and CeRu_2Si_2 (3–4).⁷⁰ Therefore, DFT+DMFT calculations are necessary to describe the electronic structure properly. The electronic specific heat coefficient, γ , was obtained theoretically from the density of

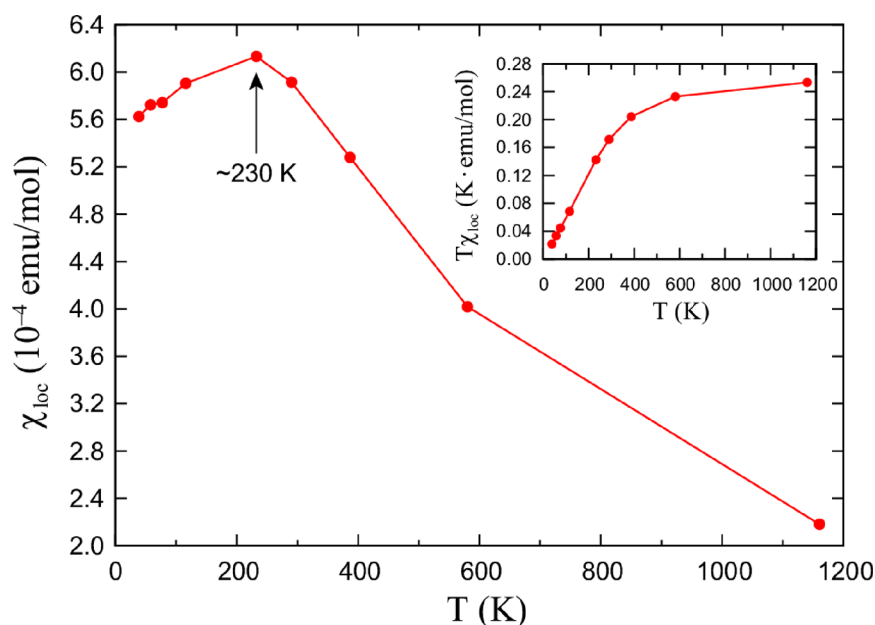


Figure 5. Temperature-dependent local magnetic susceptibility computed by DFT+DMFT. Inset shows the temperature-dependent $T\chi_{\text{loc}}$ curve. It follows the Curie–Weiss law at a very high temperature of ~ 1000 K.

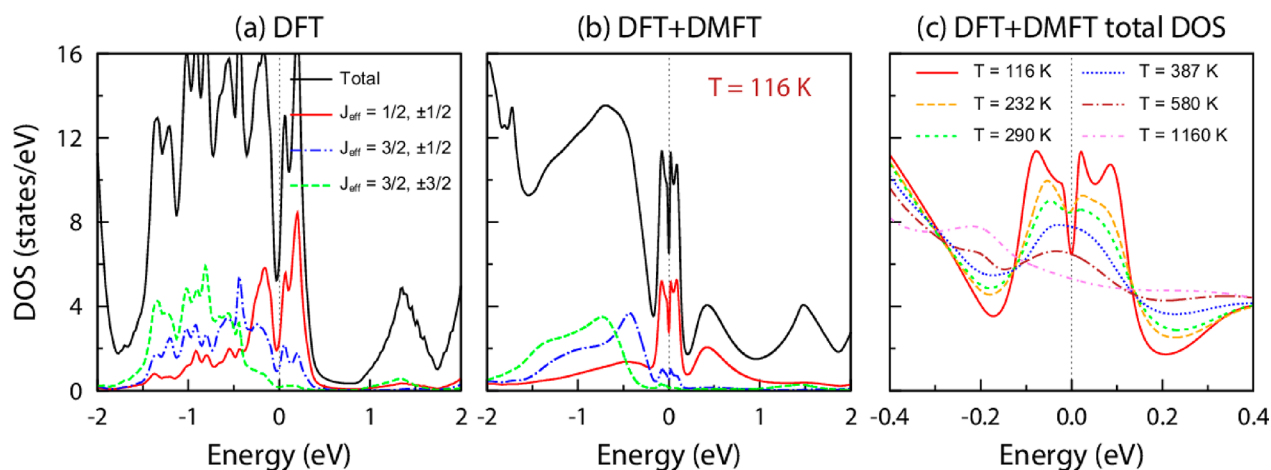


Figure 6. Density of states (DOS) of $\text{Tl}_2\text{Ir}_2\text{O}_7$ computed by (a) DFT and (b) DFT+DMFT at $T = 116$ K. Orbital-resolved DOS is provided in (a) and (b). (c) Temperature-dependent DOS computed by DFT+DMFT. Dip feature is realized in DFT and DFT+DMFT at low temperature.

states at the Fermi level, 6.80 and $9.99 \text{ mJ mol}^{-1} \text{ K}^{-2}$, respectively, for DFT and DFT+DMFT (at $T = 290$ K). The local magnetic susceptibility, χ_{loc} , is computed in the CTQMC impurity solver by using the formula below:⁷¹

$$\chi_{\text{loc}} = \frac{\mu_0 (g\mu_B)^2 N_A}{k_B} \int_0^\beta d\tau \langle J_z(\tau) J_z(0) \rangle$$

where J_z , μ_0 , g , μ_B , N_A , k_B , and β are the total angular momentum of the Ir atom, the vacuum permeability, the Landé g -factor, the Bohr magneton, the Avogadro number, the Boltzmann constant, and the inverse temperature, respectively.

Figure 5 shows the computed χ_{loc} as a function of temperature. The computed χ_{loc} at $T = 230$ K is $\chi_{\text{loc}} = 6.14 \times 10^{-4} \text{ emu/mol}$, which is quite comparable to the experimental one at the same temperature. Therefore, DFT+DMFT produces both experimental values of γ and χ almost correctly, thereby affirming that DFT+DMFT explains well the experimental data. The computed χ_{loc} follows the Curie–Weiss

law at a very high temperature (~ 1000 K), which is attributed to the local magnetic moment of the Ir atom, and it is estimated to be $\sim 0.6 \mu_B$ per Ir atom. It is noteworthy that the computed χ_{loc} is nearly saturated below $T \sim 230$ K, where a dip feature begins to appear in the DFT+DMFT total DOS (see Figure 6c).

According to the ionic model, $\text{Tl}_2\text{Ir}_2\text{O}_7$ has Ir^{4+} , which corresponds to the d^5 electronic configuration. The Ir atoms are coordinated by six O atoms forming octahedra, as a result, the Ir 5d orbitals are split into t_{2g} and e_g orbitals; the t_{2g} orbital is further split into the effective J basis of $J_{\text{eff}} = 1/2$ and $3/2$, because of the strong SOC.¹ From the simple ionic picture, $J_{\text{eff}} = 3/2$ (degeneracy: 4) is fully occupied, but $J_{\text{eff}} = 1/2$ (degeneracy: 2) is half-filled, thereby the Mott physics is relevant in the $J_{\text{eff}} = 1/2$ basis.

Figure 6a,b shows the density of states (DOS) computed by DFT and DFT+DMFT at $T = 116$ K, respectively, where a dip feature is realized in both DFT and DFT+DMFT calculations. Partial DOSs for $J_{\text{eff}} = 1/2$ and $3/2$ are also provided, and their

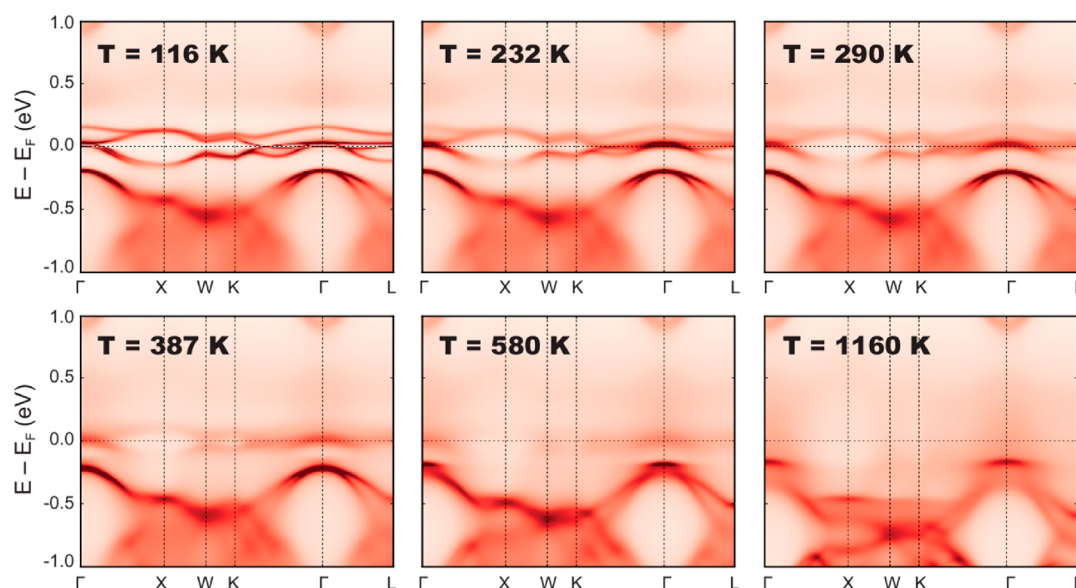


Figure 7. Temperature-dependent band dispersion of $\text{Tl}_2\text{Ir}_2\text{O}_7$ computed by DFT+DMFT.

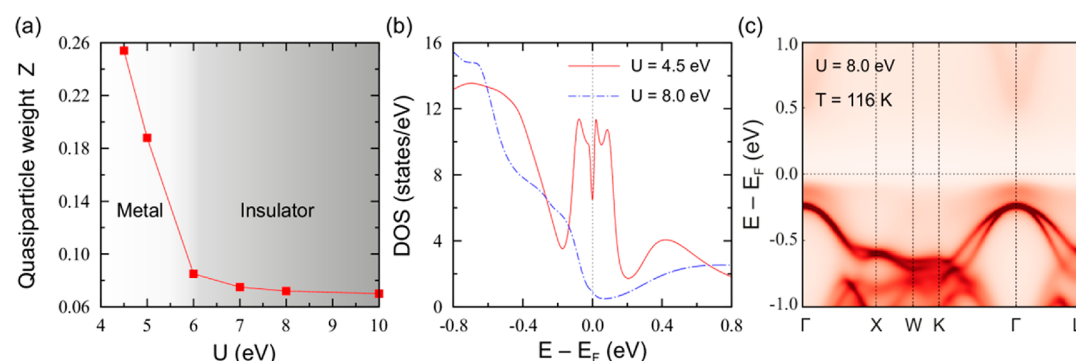


Figure 8. Electronic correlations in $\text{Tl}_2\text{Ir}_2\text{O}_7$ as a function of on-site Coulomb repulsion U (at a fixed $T = 116$ K) computed by DFT+DMFT. (a) Quasi-particle weight Z of $J_{\text{eff}} = 1/2$ as a function of U . Metal-to-insulator phase transition occurs at around $U = 6$ eV. (b) Density of states (DOS) calculated with different U . (c) Band dispersion at $U = 8$ eV. Both DOS and band dispersion clearly show an insulating phase at $U = 8$ eV.

occupation numbers are 1.25 (1.06) and 3.71 (2.87), respectively, in the DFT+DMFT (DFT) calculations. In the DFT+DMFT calculations, $J_{\text{eff}} = 3/2$ is almost fully occupied, and $J_{\text{eff}} = 1/2$ is nearly half-filled, whereby the Mott physics is relevant in the $J_{\text{eff}} = 1/2$ band. The bandwidth of $J_{\text{eff}} = 1/2$ is highly renormalized due to its strong correlation effect (the quasi-particle weight Z estimated from the self-energy is 0.25 at $T = 116$ K) as displayed in Figure 6b. It indicates that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is close to a MIT driven by the Mott physics. It is supported by the DFT+DMFT studies on $\text{Pr}_2\text{Ir}_2\text{O}_7$, which has almost the same bandwidth and is at the border of a MIT.⁵⁴

Temperature-dependent DOSs and band dispersions are displayed in Figures 6c and 7, respectively. Upon cooling, $J_{\text{eff}} = 1/2$ becomes coherent and has a larger spectral weight at the Fermi level. However, the dip feature is realized at low temperature, $T = 116$ K, and gives a small spectral weight at the Fermi level as shown in Figure 6c. The strong electronic correlations in $J_{\text{eff}} = 1/2$, indicate strong temperature dependence of the spectral weight near the Fermi level, which indicates that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is close to a MIT and might explain the resistivity upturn at low temperatures.

The fact that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is close to a MIT is further supported by Figure 8 where electronic correlations were studied as a function of on-site Coulomb repulsion U . The critical U value

for a MIT is around 6 eV, and an insulating phase is realized above the critical value as shown in Figure 8. Hence, first-principles calculations clearly demonstrate that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is proximal to a MIT.

DISCUSSION

Studies of iridium pyrochlores oxides, $\text{A}^{3+}_2\text{Ir}^{4+}_2\text{O}_7$ ($A =$ lanthanide ion, Bi, and Y), indicate that the electrical properties are correlated to the ionic size of A^{3+} ions which determine the trigonal distortions of IrO_6 and Ir–O–Ir bond angles, thus affecting the hopping between neighboring Ir electrons.^{9,32} $\text{Bi}_2\text{Ir}_2\text{O}_7$ and $\text{Pr}_2\text{Ir}_2\text{O}_7$ with relatively large A^{3+} cations are metallic,^{10,16,19,25} while others with smaller A^{3+} ions are antiferromagnetically ordered and electrically insulating at low temperatures;^{17,18,21,27,31} particularly, $\text{Nd}_2\text{Ir}_2\text{O}_7$, $\text{Sm}_2\text{Ir}_2\text{O}_7$, and $\text{Eu}_2\text{Ir}_2\text{O}_7$ show a MIT.^{12,21,28–30} The ionic radius of Tl^{3+} with TlO_8 coordination as given by Shannon is 0.98 Å which is much smaller than that of Pr^{3+} (1.126 Å) but is comparable to those of Yb^{3+} (0.985 Å) and Lu^{3+} (0.977 Å).³³ Thus, $\text{Tl}_2\text{Ir}_2\text{O}_7$ would be expected to be insulating, because both $\text{Yb}_2\text{Ir}_2\text{O}_7$ and $\text{Lu}_2\text{Ir}_2\text{O}_7$ are insulating.

In this work, the polycrystalline sample of $\text{Tl}_2\text{Ir}_2\text{O}_7$ has been successfully synthesized. The Ir^{4+} valence state is confirmed by the Ir– L_3 XANES spectrum. Magnetic susceptibility data

indicate that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is Pauli paramagnetic. Temperature-dependent resistivity and the low-temperature specific heat data support the metallic state of $\text{Tl}_2\text{Ir}_2\text{O}_7$, though the resistivity shows an upturn at low temperatures. DFT+DMFT calculations also support the metallic state of $\text{Tl}_2\text{Ir}_2\text{O}_7$ and show the strong electronic correlations, which may be responsible for the resistivity upturn and indicates that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is proximal to a MIT.

Both our experimental and theoretical studies show that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is not an insulator, but it is a Pauli paramagnetic metal and is close to a MIT. Thus, $\text{Tl}_2\text{Ir}_2\text{O}_7$ is uniquely different from the established phase diagram of the pyrochlore iridate oxides $\text{A}^{3+}_2\text{Ir}^{4+}_2\text{O}_7$.⁵⁴ It should be noted that Takeda et al. suggested that the Ti^{3+} ionic radius of TlO_8 coordination in pyrochlore $\text{Tl}_2\text{Ru}_2\text{O}_7$ should be around 1.08–1.09 Å.⁵⁶ From the average Ti – O bond distance (2.420 Å) of $\text{Tl}_2\text{Ir}_2\text{O}_7$, a Ti^{3+} effective ionic radius of 1.02 Å with TlO_8 coordination was suggested. However, even with this increased Ti^{3+} ionic size correction, it is still not sufficient to justify the observed metallic state of $\text{Tl}_2\text{Ir}_2\text{O}_7$, as our experimental and theoretical studies indicate that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is distinctive from the current $\text{A}^{3+}_2\text{Ir}^{4+}_2\text{O}_7$ phase diagram.

Figure 9 displays that the Ir – O – Ir bond angles versus the effective ionic radius of $\text{Tl}_2\text{Ir}_2\text{O}_7$ and $\text{In}_2\text{Ir}_2\text{O}_7$ do not follow

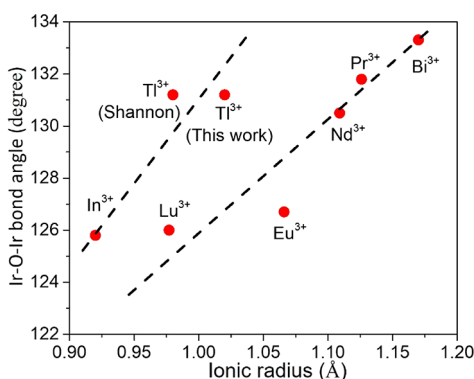


Figure 9. Ir – O – Ir bond angle of $\text{A}^{3+}_2\text{Ir}^{4+}_2\text{O}_7$ vs effective ionic radius of A^{3+} . The 8-coordinated Ti^{3+} ionic radius was taken both from Shannon,³³ and the one derived from the average Ti – O bond length of this work. The other ionic radii were taken from Shannon.³³ The dashed lines are just for the eyes.

the trend of the other rare-earth iridium pyrochlores, $\text{A}_2\text{Ir}_2\text{O}_7$, and have relatively larger Ir – O – Ir bond angles. The uniqueness of $\text{Tl}_2\text{Ir}_2\text{O}_7$ and $\text{In}_2\text{Ir}_2\text{O}_7$ may be attributed to the nature of bonding. Ti^{3+} ($4f^1 4d^5 6s^0 6p^0$) and In^{3+} ($4d^{10} 5s^0 5p^0$) ions have fully filled f and d orbitals, whereas the trivalent rare earth ions have $4f^n 5d^0 6s^0$ configurations (n varies from 1 for Ce^{3+} to 14 for Lu^{3+}). The fully occupied d orbitals for Ti^{3+} and In^{3+} would be far below the Fermi energy and could hinder their hybridization with the surrounding oxygens. The empty $5d$ orbitals of rare-earth ions would enhance their hybridization with the surrounding oxygens.

The Ir – O – Ir bond angle 131.2° of $\text{Tl}_2\text{Ir}_2\text{O}_7$ is between those of 131.8° for metallic $\text{Pr}_2\text{Ir}_2\text{O}_7$ ²² and 130.50° for $\text{Nd}_2\text{Ir}_2\text{O}_7$ with MIT.¹⁴ The Ir – O – Ir bond angle determines the overlapping of Ir – O orbitals and governs the hopping between the neighboring Ir $5d$ electrons and the physical properties.⁹ The Ir – O – Ir bond angle of $\text{Tl}_2\text{Ir}_2\text{O}_7$ is consistent with our experimental and the DFT+DMFT calculation results that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is proximal to a MIT. These findings indicate that

the electronic configurations of the A^{3+} ions significantly affect the crystal structure of iridium pyrochlore oxides and the corresponding electrical and magnetic properties.

CONCLUSIONS

Pyrochlore iridate $\text{Tl}_2\text{Ir}_2\text{O}_7$ was synthesized under high-pressure conditions. $\text{Tl}_2\text{Ir}_2\text{O}_7$ crystallizes in the cubic pyrochlore crystal structure with space group $Fd\bar{3}m$ (No. 227). The Ir^{4+} oxidation state is confirmed by the Ir – L_3 XANES spectrum. The ionic size of Ti^{3+} is much smaller than that of Pr^{3+} ; hence, $\text{Tl}_2\text{Ir}_2\text{O}_7$ would be expected to be insulating according to the established phase diagram of the pyrochlore iridate compounds $\text{A}^{3+}_2\text{Ir}^{4+}_2\text{O}_7$.⁹ However, combined temperature-dependent magnetic susceptibility, resistivity, specific heat, and DFT+DMFT calculation data show that $\text{Tl}_2\text{Ir}_2\text{O}_7$ is a Pauli paramagnetic metal, and it is close to a MIT. This uniqueness of $\text{Tl}_2\text{Ir}_2\text{O}_7$ is attributed primarily to the electronic configuration difference of Ti^{3+} relative to the rare-earth ions, which plays a substantial role in determining the Ir – O – Ir bond angle, and the correlated electrical and magnetic properties. $\text{Tl}_2\text{Ir}_2\text{O}_7$ at the brink of localization provides a new platform to study and understand the MIT.

ASSOCIATED CONTENT

Accession Codes

CCDC 2039823 contains the supplementary crystallographic data for this paper. These data can be obtained free of charge via www.ccdc.cam.ac.uk/data_request/cif, or by emailing data_request@ccdc.cam.ac.uk, or by contacting The Cambridge Crystallographic Data Centre, 12 Union Road, Cambridge CB2 1EZ, UK; fax: +44 1223 336033.

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Notes

The authors declare no competing financial interest.

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