

Light Induced Processes in CsPbBr₃–Au Hybrid Nanocrystals: Electron Transfer and Expulsion of Au

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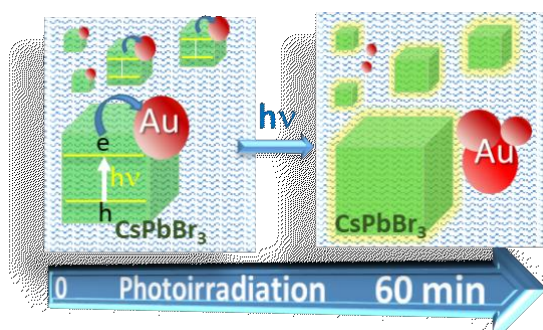
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Abstract

Semiconductor-metal heterostructures such as $\text{CsPbBr}_3\text{-Au}$ are useful in photocatalysis. When Au nanoparticles are deposited on the CsPbBr_3 nanocrystal surface they efficiently quench the photoluminescence of the semiconductor. This process has been studied by femtosecond transient absorption spectroscopy measurements, which indicate that electron transfer to the Au nanoparticles occurs from both hot and relaxed electrons in the conduction band of CsPbBr_3 . The electron transfer rate constant is much larger for the hot electrons compared to the relaxed electrons. Under steady state photoirradiation of $\text{CsPbBr}_3\text{-Au}$ heterostructure, the photogenerated electrons from the excited CsPbBr_3 nanocrystals continue to charge the Au nanoparticles. After sufficient irradiation, the gold nanoparticles dissociate from the CsPbBr_3 surface and aggregate into larger size gold nanoparticles. The expulsion of gold nanoparticles restores the original luminescence behavior of CsPbBr_3 nanocrystals. The spectroscopic and morphological studies provide insight into the expulsion of gold nanoparticles in photoirradiated $\text{CsPbBr}_3\text{-Au}$ heterostructures.

TOC Graphics



Introduction

Semiconductor-metal heterostructures are widely used in photocatalytic water splitting and other chemical transformations.¹⁻³ The presence of a noble metal (e.g. Au) at the semiconductor surface can quickly capture the photogenerated electrons and thus facilitates overall charge separation.⁴ The photocatalytic performance of such a noble metal-modified semiconductor can lower the overpotential for reduction of H⁺ ions. While recent focus involving gold nanoparticles has centered around plasmonic catalysis or hot electron transfer, they are inherently good electron acceptors and promote reduction processes.⁵⁻¹³ The electron storage and transfer properties of gold nanoparticles can be realized through nonplasmonic (<3 nm) Au nanoparticles. Electron storage in small Au nanoparticles has been demonstrated through quantized capacitance measurements.^{14,15} Blue-shifts in the plasmon band of the metal nanoparticles during photoirradiation of semiconductor-metal heterostructures, such as TiO₂-Ag or TiO₂-Au, have also been used to demonstrate electron transfer from the semiconductor to the metal.¹⁶⁻¹⁸

The emergence of CsPbBr₃ nanocrystals, a nanoscale metal halide perovskite system with high luminescent and favorable charge transfer properties offers new opportunities to explore applications of perovskites in photocatalysis.¹⁹⁻²¹ The ability to tune their bandgap through halide composition makes these materials potential candidates for the development of solar cells,²²⁻²⁵ light emitting devices^{26,27} and high energy detectors.²⁸⁻³¹ Plasmonic gold nanoparticles have been employed to boost the performance of perovskite solar cells.³²⁻³⁴ However, the diffusion of gold from the metal contact in a solar cell has been found to deteriorate the solar cell performance.^{35,36}

Halide perovskites are inherently unstable in polar solvents. Although efforts have been made to design core@shell heterostructures such as CsPbBr₃@ZnS³⁷ and CsPbBr₃@CdS³⁸ to improve their stability in polar solvents, their application in photocatalysis remain limited. Another difficulty in designing perovskite photocatalyst is the synthesis of CsPbBr₃-metal heterostructure with a controllable morphology. In our earlier study, we showed that addition of small amounts of AuBr₄⁻ resulted in the formation of nanometer sized gold nanoparticles.³⁹ If we increase the concentration of AuBr₄⁻ to grow larger size plasmonic Au nanoparticles, one sees substitution of Pb²⁺ with Au³⁺ to form Cs₂Au₂Br₆.⁴⁰ This substitution is driven by the concentration of metal ions and can be readily suppressed in the presence of excess Pb²⁺ ions. Thus, the design of a stable and tunable perovskite-metal heterostructures is a complicated process that remains a major challenge in perovskite photocatalysis.

The obvious question then is why it is so difficult to design a perovskite-metal nanoassembly that can be directly employed in a photocatalysis process. This paper addresses this issue through studies of CsPbBr₃-Au heterostructures. Spectroscopic and morphological changes following bandgap excitation of the CsPbBr₃ moiety have been examined. The results show fast electron transfer from CsPbBr₃ to Au, and subsequent photoinduced expulsion of the Au nanoparticles from the CsPbBr₃-Au heterostructures.

Experimental Section

Chemicals. Cesium carbonate (Cs₂CO₃, 99.9%, Sigma Aldrich), Lead (II) bromide (PbBr₂, 99.99%, Sigma Aldrich), Oleic acid (OLA, 90%, Sigma Aldrich), Oleylamine (OLAm, technical grade 70%, Sigma Aldrich), 1-Octadecene (ODE, technical grade 90%, Sigma Aldrich), Methyl acetate (MeOAc, 99.98%, Gold(III) bromide (AuBr₃, 99.99%, Sigma Aldrich), Toluene (Anhydrous, 99.99%, Sigma Aldrich), Ethanol (200 proof, Koptec).

Synthesis of CsPbBr₃ nanocrystals (NCs). CsPbBr₃ NCs were synthesized following a modified procedure adapted from Protesescu et al.⁴¹ Cs-oleate was prepared by mixing 0.153 g of Cs₂CO₃, 1.4 mL of OLA, and 3 mL of ODE in a 25 mL round bottom flask. The mixture was degassed at 80 °C for 1 hour. The lead bromide precursor was prepared by adding 3 mL of OLA, 3 mL of ODE, and 0.414 g of PbBr₂ to a 25 mL round bottom flask and degassing for 1 hour at 80 °C. After one hour, the temperature was raised to 170 °C, and 1.5 mL of OLAm was added to the mixture. The solution was degassed for one more minute. The vacuum was then turned off, and the flask was put into a nitrogen environment. Simultaneously, the Cs-oleate precursor was brought up to 120 °C and put under a nitrogen atmosphere. Once both solutions have reached the target temperatures, 2 mL of the Cs-oleate solution was injected into the lead-bromide precursor with a heated syringe (from the oven). The mixture was then immediately put in an ice bath until it reached a temperature of 65 °C.

The nanocrystals were washed with approximately 20 mL of 1-Octadecene and transferred to a 50 mL centrifuge tube. This was then spun down for 10 minutes at 7800 rpm. The supernatant was discarded, and the pellet was washed with 5 mL of ODE. The nanocrystals were again centrifuged for 10 minutes at 7800 rpm. The pellet was dispersed in 5 mL toluene and washed with 5 mL of methyl acetate. The nanocrystals were spun down again for 10 minutes at 7800 rpm. The supernatant was discarded, and the pellet was dispersed in 3 mL toluene and again centrifuged for 10 minutes at 7800 rpm. The supernatant was collected and preserved for further experiments.

Synthesis of Au-CsPbBr₃ hybrid nanocrystals. AuBr₃ was dissolved in 1:10 v/v ethanol/toluene mixture to get 5.15 mM solution. 10 µL OLA was added to 1 mL of 5.15 mM AuBr₃ solution and the solution was never stored for longer than an hour.⁴⁰ The prepared AuBr₃ solution was swiftly added to CsPbBr₃ nanocrystals in toluene to get the hybrid nanocrystals. The stability of CsPbBr₃ in 0.1 and 0.4% ethanol containing toluene was checked by monitoring absorption and emission spectra (Figure S1). It was found to be stable.

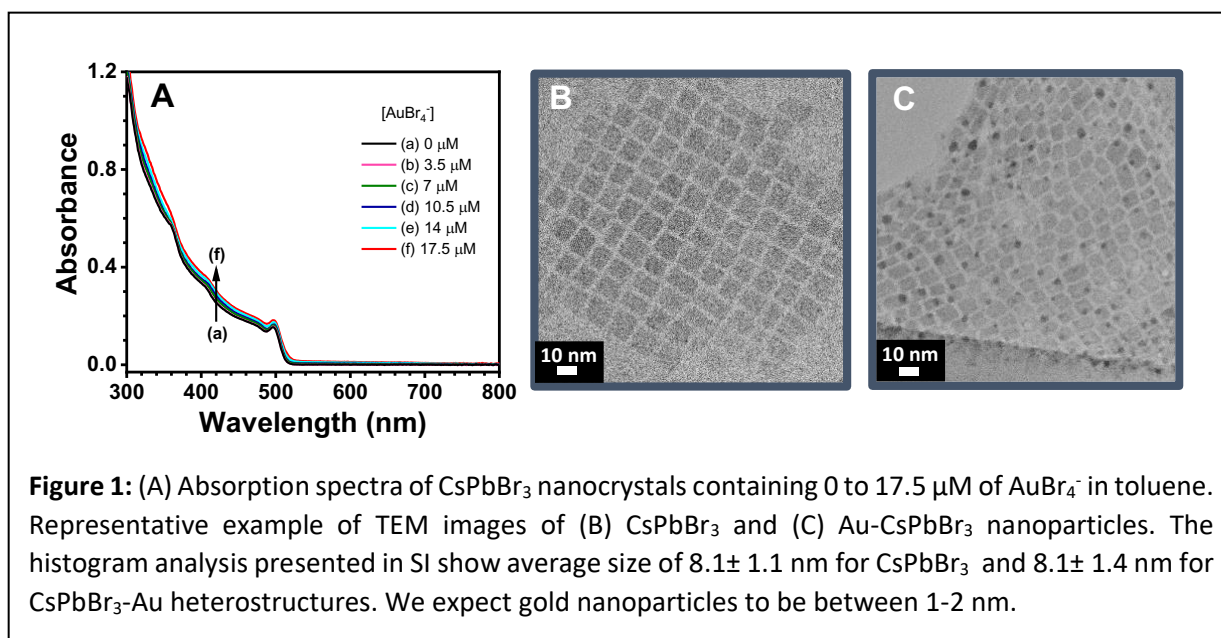
Characterization Methods. UV-visible absorption spectra were recorded using a Cary 50 Bio spectrophotometer (Varian). Emission spectra were collected using a Horiba Jobin Yvon Fluorolog-3 spectrophotometer. The photoluminescence decay of the samples was monitored and recorded by time-correlated single-photon counting (TCSPC) using a Horiba Jobin Yvon system with a IBH DataStation Hub for timing with a 371nm nano LED light source. Transmission electron microscopy (TEM) and Scanning Transmission Electron Microscopy (STEM) imaging were carried out using FEI Titan 80-300 (300 kV) high-resolution transmission electron microscope. X-ray diffraction (XRD) analysis was performed by using D8 Advance Davinci, Bruker X-ray diffractometer with Cu K α X-ray ($\lambda = 1.5406 \text{ \AA}$) and for all the samples, the value of 2θ was varied from 10° to 70° with a scan rate of 0.3° min⁻¹. The colloidal suspension of nanocrystals in a quartz cuvette was subjected to 1Sun (100 mW/cm²) visible light excitation. The absorbance and emission spectra and other spectral characterization were made on samples irradiated at different irradiation times.

Transient Absorption Experiments. Femtosecond transient absorption measurements were carried out using a Spectra-Physics Solstice Ace laser system (800 nm fundamental, ~5 mJ/pulse, fwhm = ~30 fs, repetition rate of 1 kHz) with detection software from Ultrafast Systems (Helios). A small portion of the fundamental beam was frequency-doubled to 400 nm (using a 1 mm thick BBO crystal) to generate the pump beam, and another part of the fundamental beam was passed through an optical delay stage and focused onto a sapphire crystal to generate a white-light continuum probe pulse. The pump and probe

beams are spatially overlapped onto the sample such that the probe is within the area of pump irradiation. The pump beam was sent through an iris of known diameter and the power was adjusted via a neutral density filter wheel to keep the power density at $\sim 15 \mu\text{J}/\text{cm}^2$ for these experiments. Under these conditions the transient absorption experiments do not show any evidence of multiple exciton/Auger recombination effects. The 2D perovskite films were placed in a custom vacuum cell and pumped down/refilled with argon 3x before measurement, to eliminate any potential interactions with oxygen.

Results and Discussion

CsPbBr₃ nanocrystals and CsPbBr₃-Au Heterostructures. The CsPbBr₃ nanocrystal suspension stabilized with oleylamine/oleic acid ligands was prepared as per the literature procedure.⁴¹ Spectrum *a* in Figure 1A shows the absorption characteristics of CsPbBr₃ nanocrystals with an excitonic peak at 500 nm. Addition of AuBr₄⁻ to this solution results in the formation of Au nanoparticles on the surface of the

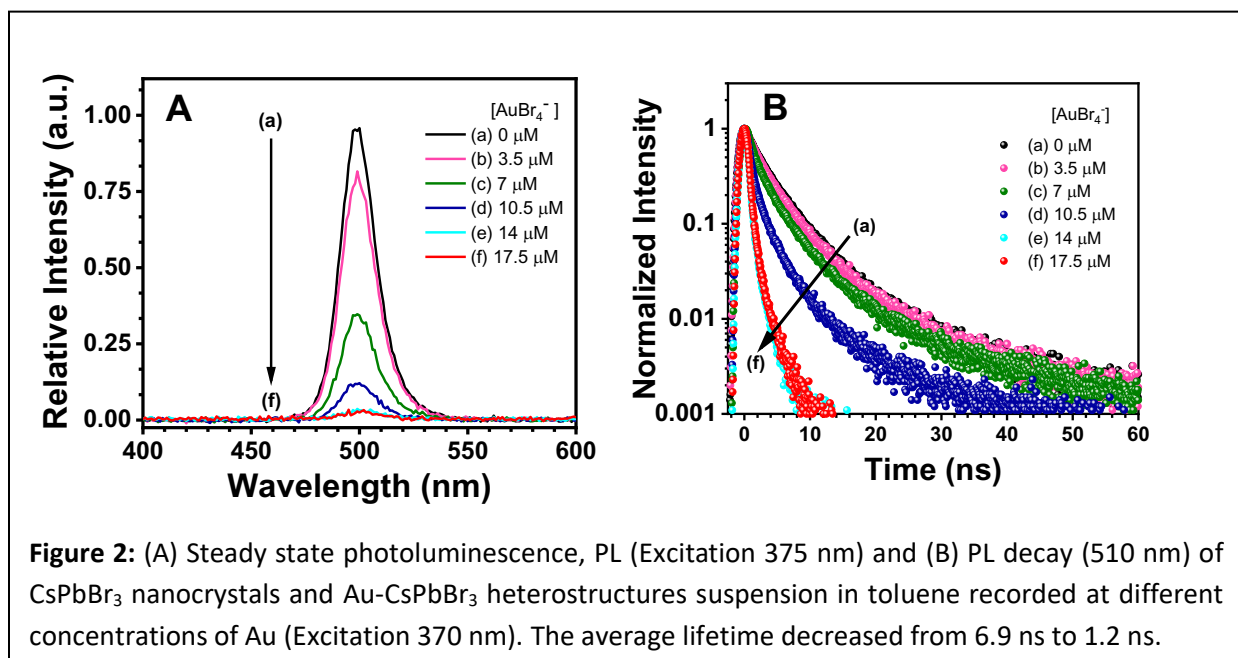


perovskite nanocrystals. The amine groups of oleylamine are capable of reducing AuBr₄⁻ to form small Au nanoparticles.^{40,42} Representative TEM images of the CsPbBr₃ and CsPbBr₃-Au samples are shown in Figure 1. The gold nanoparticles formed on the surface of CsPbBr₃ have diameters in the range of 1-2 nm (Figure S2C in the supporting information). Size distribution analysis (Figure S2A and S2B in the supporting information) shows average sizes of the semiconductor moiety of 8.1 ± 1.1 nm for CsPbBr₃ and 8.1 ± 1.4 nm for CsPbBr₃-Au nanocrystals. This shows that there are no major changes in the morphology of the CsPbBr₃ nanocrystals upon the deposition of the Au nanoparticles onto their surfaces. Based on the TEM image analysis we expect $\sim 85\%$ of CsPbBr₃ particles to have at least one Au nanoparticle. (Any Au clusters that may be hidden in the background of CsPbBr₃ are not included in this coverage) The gold nanoparticles formed on the surface of CsPbBr₃ in our experiments are too small to exhibit a plasmon resonance. However, their presence is confirmed through the small increase in the absorption below 500 nm and the TEM measurements (Figure 1C). The XRD pattern of CsPbBr₃ and CsPbBr₃-Au nanoparticles are shown in

Figure S3. The XRD pattern CsPbBr₃-Au shows an additional peak corresponding to Au, thus confirming its presence in the hybrid nanostructure.

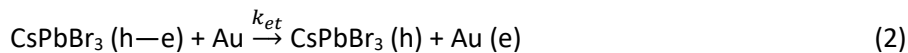
It has been shown earlier that Au(III) at higher concentrations can substitute Pb²⁺ to form Cs₂Au₂Br₆.⁴⁰ This substitution is marked by the strong absorbance in the long wavelength region (500-700 nm). The absence of such absorbance in the long wavelength region indicates that the concentration range of <17.5 μ M of AuBr₄⁻ employed in the present study excludes the possibility of the Au for Pb substitution. One needs to increase the concentration of gold salt to the millimolar range to see such effects.

Tracking Excited State Interactions. The CsPbBr₃ nanocrystals are highly luminescent with a quantum efficiency reaching near unity.⁴³ The bright green emission of CsPbBr₃ provides a convenient means to probe excited state interaction with the small gold nanoparticles. The emission spectra of CsPbBr₃



nanocrystals in toluene are shown in Figure 2A. With increasing concentration of AuBr₄⁻ we observe quenching of CsPbBr₃ emission. Nearly 100% quenching is seen at a concentration of 17.5 μ M AuBr₄⁻. As shown earlier, at these low concentrations AuBr₄⁻ interacts with the oleylamine to form gold nanoparticles. These gold nanoparticles are homogeneously distributed across different CsPbBr₃ nanoparticles, mostly at the corner sites.

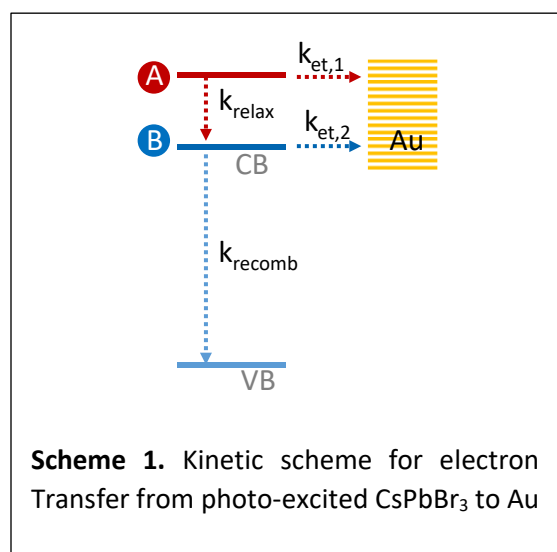
The quenching of emission can be understood from the ability of gold nanoparticles to serve as electron acceptors. These nonplasmonic metal nanoparticles have been shown to capture and store electrons from excited semiconductor nanoparticles.⁴⁴⁻⁴⁶ The excited state deactivation can be understood from reactions 1 and 2.



As reaction 2 competes with the reaction 1, we see a decreased emission from the CsPbBr₃ nanocrystals when Au is added. We also monitored the emission decay of CsPbBr₃ at different Au concentrations. Figure 2B shows emission decay recorded at 510 nm using 370 nm pulsed laser excitation. The decay curves were analyzed using a biexponential fit, $I(t) = \sum_{i=1}^n A_i \exp(-t/\tau_i)$. The longer component is likely to arise from the CsPbBr₃ nanocrystals that may not have a coverage of Au. In fact with increasing Au concentration, this long lifetime component also decreases. The preexponent factors and lifetimes are presented in Table S1. The two lifetimes estimated for the emission decay decreased from 3.1 ± 0.02 ns and 9.7 ± 0.06 ns for pristine CsPbBr₃ to 0.3 ± 0.004 ns and 2.9 ± 0.01 ns for CsPbBr₃-Au (corresponding to 17.5 μ M Au). The corresponding average lifetime, $\langle\tau\rangle$, decreased from 6.9 ± 0.4 ns to 1.2 ± 0.06 ns. The decreased lifetime with increasing Au concentration supports the electron transfer quenching of excited CsPbBr₃ by the gold nanoparticles (reaction 2).

To further assess the kinetics of electron transfer, we carried out femtosecond transient absorption measurements. The transient absorption spectra of CsPbBr₃ and CsPbBr₃-Au nanocrystal suspensions were recorded at different probe delays using 400 nm laser pulse excitation (Figures 3A&B). The transient spectrum recorded immediately after laser pulse excitation of CsPbBr₃ shows intense bleaching of the exciton band, and a small induced-absorption on the red-side of the band. By comparison to other semiconductor nanocrystal systems, the bleach is attributed to state filling and the absorption to photoexcited carrier induced band-shifts.^{47,48} The formation of the bleach signal gives information about how quickly the excited charge carriers relax to the bottom of the conduction band/top of the valence band. In contrast, the dynamics of the bleach recovery report on the recombination of electron-hole pairs within the semiconductor nanoparticle, as well as other pathways by which these charge carriers are lost.⁴⁷ For example, any electron transfer to the gold nanoparticles from excited CsPbBr₃ (reaction 2) would result in faster bleach recovery.

Figures 3 C&D compare the bleach recovery recorded at 510 nm for the CsPbBr₃ and CsPbBr₃-Au systems



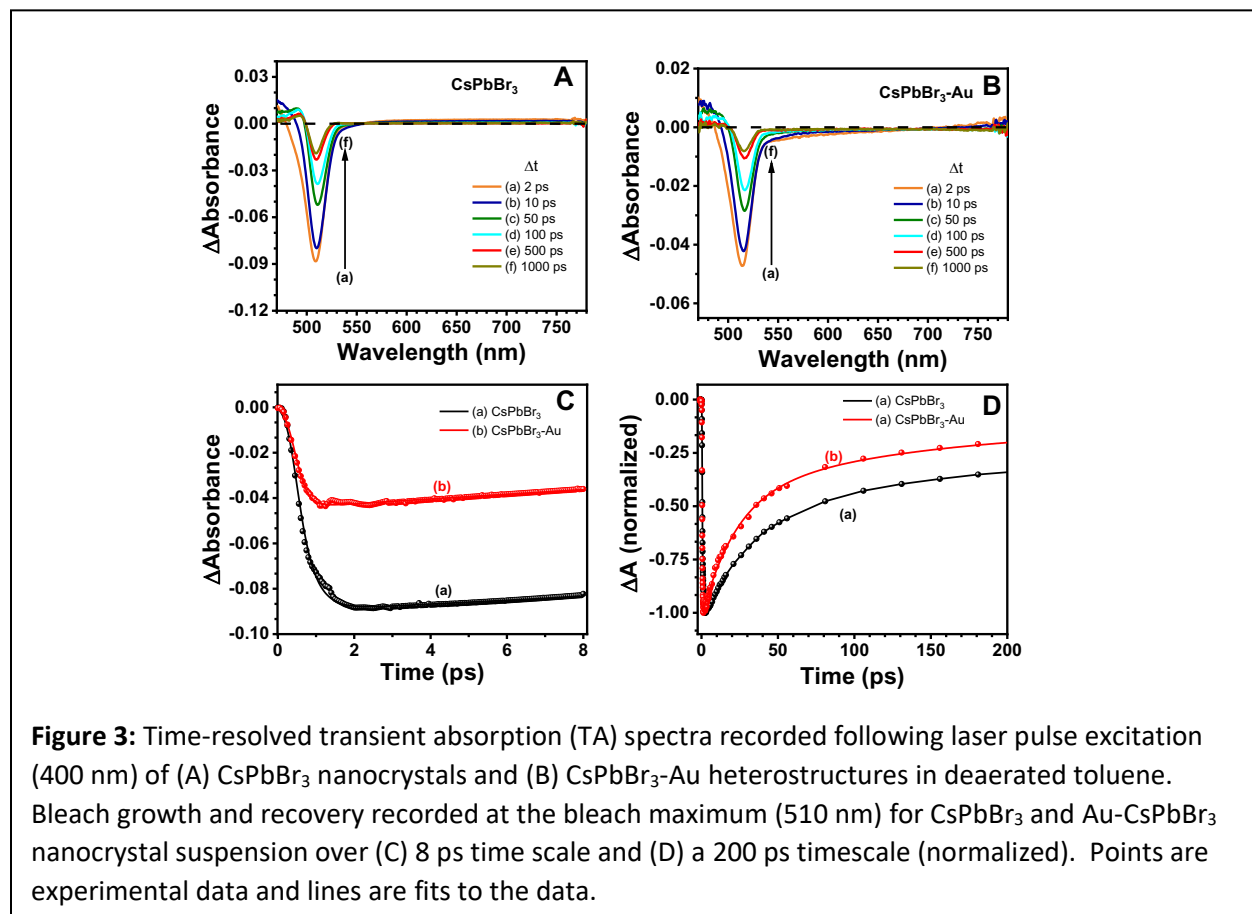
on two different times scales. The analysis of the bleach recovery data is presented in Table S2. It is noticeable from these transient absorption traces that both the rise in the bleach signal (Figure 3C) and the bleach recovery (Figure 3D) are faster for CsPbBr₃-Au compared to CsPbBr₃. The kinetic differences between the CsPbBr₃ and CsPbBr₃-Au system data are attributed to electron transfer from the CsPbBr₃ to the Au. The relatively long timescale bleach-recovery data provides information about electron transfer for electrons relaxed to the bottom of the conduction band, and fast rise time data is sensitive to electron transfer from hot electrons in the CsPbBr₃ conduction band (Scheme 1).

If we assume electron transfer from excited CsPbBr₃ to Au is the only major pathway responsible for faster kinetics, we can estimate a rate constant for electron transfer using the equation (3),

$$k_{et} = 1/\tau(\text{CsPbBr}_3\text{-Au}) - 1/\tau(\text{CsPbBr}_3) \quad (3)$$

where $\tau(\text{CsPbBr}_3\text{-Au})$ represents the lifetime for the bleach rise (or recovery) of the hybrid nanostructure and $\tau(\text{CsPbBr}_3)$ represents the lifetime of the bleach rise (or recovery) of the pristine CsPbBr₃ nanocrystals.

In the following, we analyze the bleach recovery and rise time data separately to generate electron transfer rate for the relaxed and hot electrons in the CsPbBr₃ conduction band. This approach is justified by the large difference in the time constants for the rise and decay of the bleach signal. Hot electron transfer can occur to either the Au clusters, or to traps at the surface of the particles created by the clusters. These two possibilities cannot be distinguished in the transient absorption experiments. However, the observations of cluster charging and dissociation strongly suggest that electron transfer occur to the clusters. Solution of the coupled rate equations that describe electron transfer and relaxation

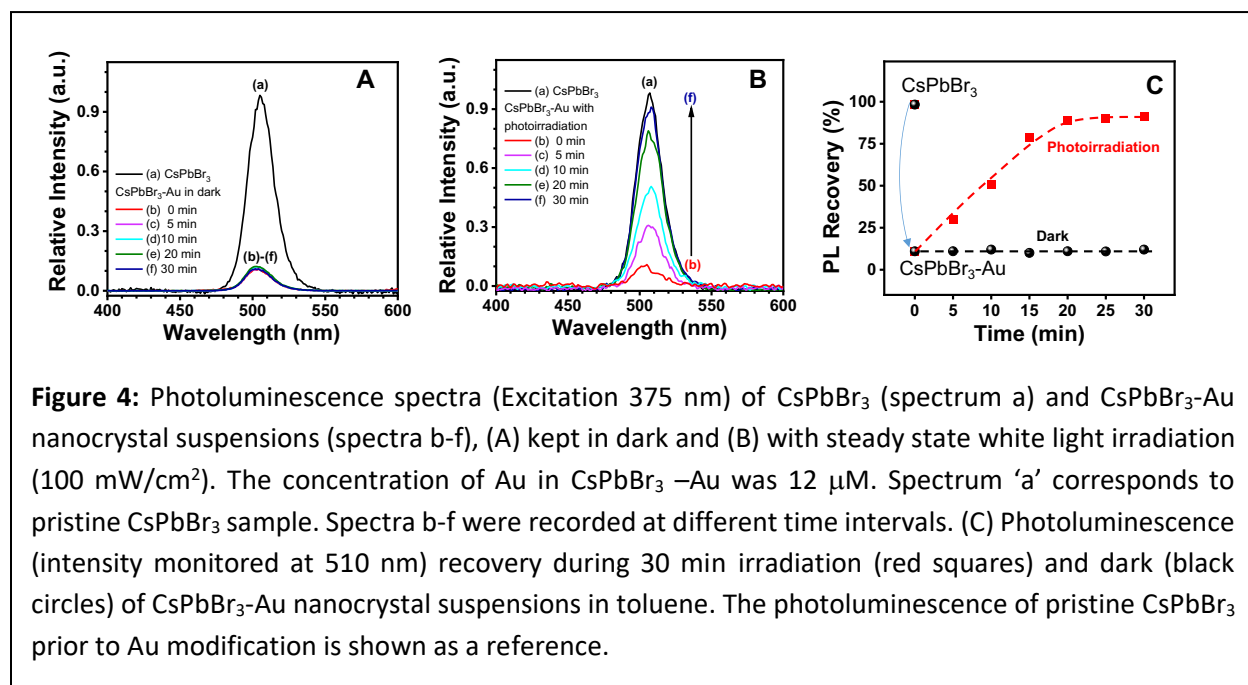


for the hot and band-edge electrons is presented in the Supporting Information. These simulations confirm that electron transfer to the Au nanoparticles from the hot electrons in the CsPbBr₃ conduction band causes a faster rise in the bleach signal.

On the 10 to 200 ps timescale, the bleach recovery time constant decreases from 36.3 ± 0.9 ps for CsPbBr₃ to 22.3 ± 1.9 ps for CsPbBr₃-Au. Substitution of these lifetimes into Equation (3) gives an electron transfer rate constant, $k_{et,2}$ of $(1.73 \pm 0.27) \times 10^{10} \text{ s}^{-1}$. This electron transfer rate is consistent with the rate constants observed for other semiconductor-metal systems for electrons relaxed to the bottom of the conduction

band.⁴⁹⁻⁵¹ On the few ps timescale the rise time for the bleach signal is 0.35 ± 0.01 ps for CsPbBr₃ compared to 0.19 ± 0.01 ps for CsPbBr₃-Au. These times constants yield an electron transfer rate constant $k_{et,1}$ of $(2.43 \pm 0.39) \times 10^{12} \text{ s}^{-1}$. The faster rate for electron transfer from the hot electrons compared to the relaxed electrons, and the magnitude of the rate for hot-electron transfer, is again consistent with previous studies of hybrid semiconductor-metal nanostructures.⁵²⁻⁵⁵ Any excess holes left in the CsPbBr₃ are scavenged by the surface ligands such as oleylamine.

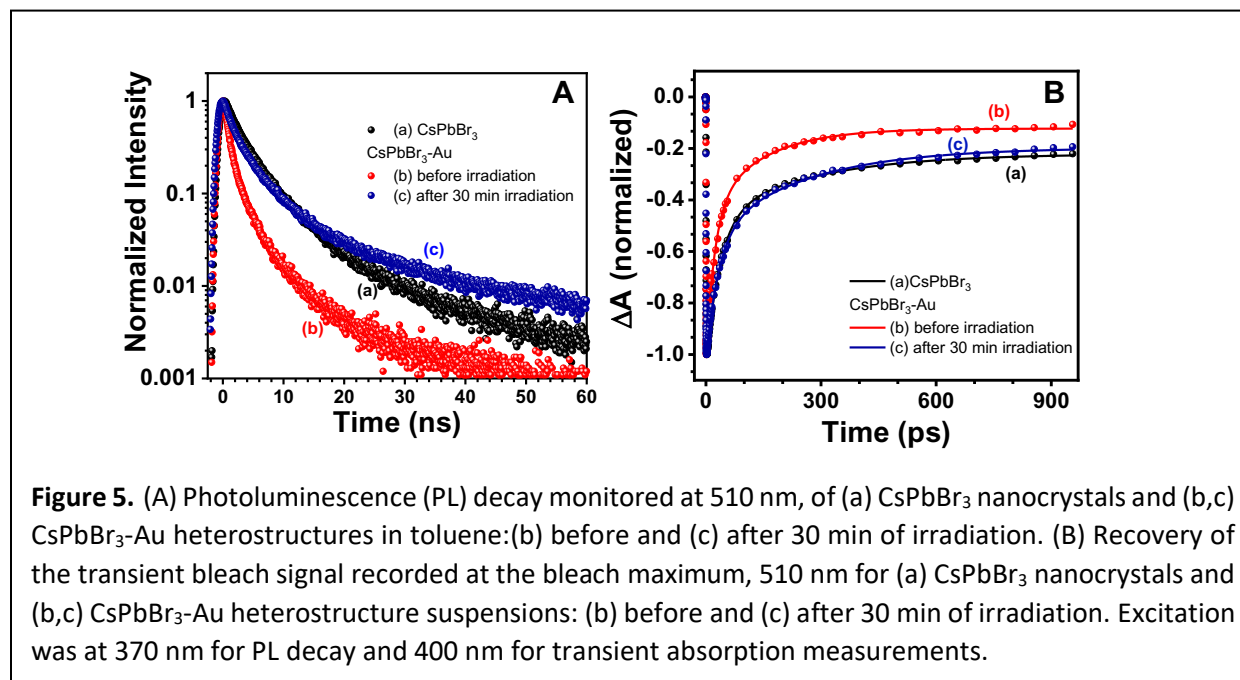
Effect of Steady-State Irradiation. One of the interesting features of perovskite nanocrystals is their potential in promoting photocatalytic reactions.^{20,56-59} The application of CsPbBr₃-metal in photocatalysis is yet to be explored fully. In order to assess the long term stability of these materials, we subjected the CsPbBr₃-Au solution to steady state irradiation using 1 Sun white light (100 mW/cm^2). In the absence of Au there is a small change in the absorption and emission spectra (Figure S4). However, the CsPbBr₃-Au samples show an interesting behavior. With continued photoirradiation the quenched emission of CsPbBr₃-Au recovers. Figure 4A and B show the emission spectra of CsPbBr₃-Au nanoparticle suspension kept in dark and under continuous white light emission. With increasing time we see an enhancement in the emission as we continue the light irradiation for 30 min. Figure 4C shows the photoluminescence recovery following the light irradiation. More than 95% of original photoluminescence was restored



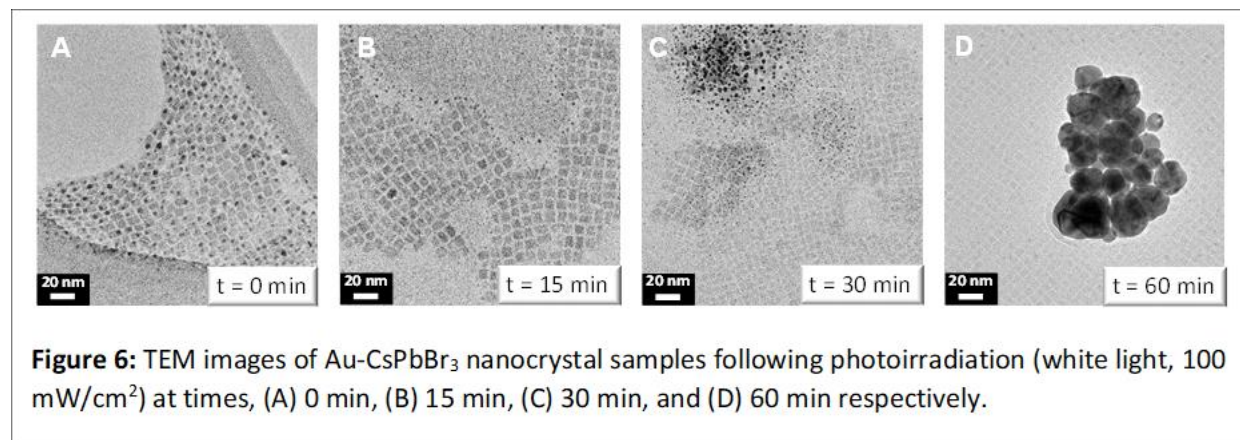
during this process. We also checked the emission recovery for CsPbBr₃-Au samples with different (1.6 μM -12 μM) Au concentrations (see SI Figures S5-S8). In all these cases, we observed a recovery of the photoluminescence (99%-85%) following steady state illumination.

The photoluminescence decay and transient bleach recovery in the transient absorption measurements are compared in Figure 5. The time resolved transient absorption spectra of photoirradiated CsPbBr₃-Au sample is shown in Figure S9. The decay profiles for photoluminescence and bleach recovery of the photoirradiated CsPbBr₃-Au sample matched that of the original CsPbBr₃ sample (i.e., without modification with gold). These results suggest that the interaction between the Au nanoparticles and the

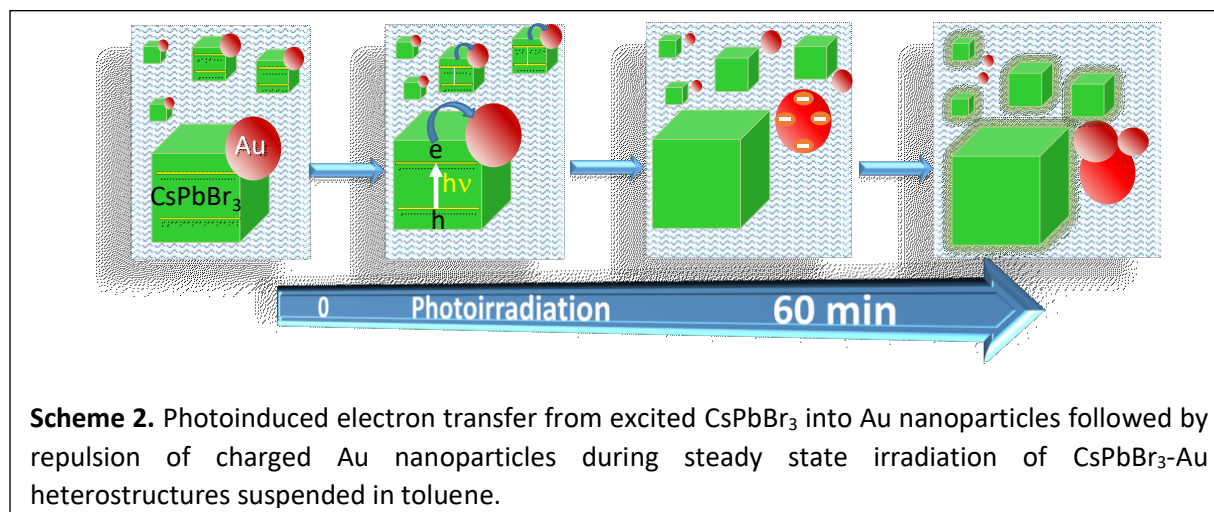
CsPbBr₃ nanocrystals has been greatly reduced following steady state irradiation with light. It is thus interesting to find out the reason behind the weakened interaction between CsPbBr₃ and Au following the photoirradiation in CsPbBr₃-Au nanocrystals.



Expulsion of Gold nanoparticles from CsPbBr₃-Au Heterostructures. We sampled the CsPbBr₃-Au nanocrystal suspension at different photoirradiation times and carried out TEM analysis. The images recorded at four different times are shown in Figure 6. Before irradiation, the gold particles are well distributed and attached to individual CsPbBr₃ nanocrystallites. Once they are subjected to photoirradiation the particles become detached (Figure 6B). At 30 minutes these small nanoparticles seem to form a network of clusters. The background image of CsPbBr₃ nanoparticles show cleaner CsPbBr₃ surfaces. With continued photoirradiation the gold clusters grow into larger size Au nanoparticles (20-100 nm diameter). These results indicate that the gold nanoparticles are no longer attached to CsPbBr₃ nanoparticles after irradiation, thus, reducing the interaction and restoring the characteristics of pristine CsPbBr₃ nanoparticles.



Semiconductor-metal heterostructures have been widely used in photocatalysis. For example, noble or precious metals deposited on a semiconductor serve as a co-catalyst to promote reduction reaction (e.g., H_2 generation in a water splitting process).⁶⁰⁻⁶² The metal particles capture electrons from the semiconductor and shuttle electrons between the semiconductor and the redox species by lowering the overpotential needed for the reduction. In the majority of these examples the anchoring of the metal particles on the semiconductor surface remains intact even during extended periods of photoirradiation. The results described in Figures 4-6 however show an unusual photo-effect of expulsion of gold nanoparticles from CsPbBr_3 -Au heterostructures. The transient absorption data shows clear evidence of photo-induced electron transfer from the CsPbBr_3 to the Au in the CsPbBr_3 -Au heterostructures. Thus, a



possible explanation for expulsion of the gold nanoparticles from the heterostructures during steady-state illumination could be charging of gold nanoparticles, which makes them unstable.

The photoinduced events of CsPbBr_3 -Au heterostructures are illustrated in Scheme 2. As shown earlier, smaller gold nanoparticles are efficient in capturing and storing photogenerated electrons from the semiconductor.^{63,64} The storage of electrons in such small metal nanoparticles has been shown by monitoring quantized capacitance,^{14,65} blue-shift in plasmon absorption¹⁶⁻¹⁸ and on-demand electron transfer at delayed time interval.⁶⁴ We expect similar electron capture by Au nanoparticles, which causes charging during photoirradiation and subsequent detachment from the CsPbBr_3 surface. Both the emission recovery and the TEM images support the steps involved in the expulsion of gold nanoparticles during photoirradiation.

Concluding Remarks

A stable design of perovskite-metal heterostructures remains a challenge. Unlike metal chalcogenide or metal oxide semiconductors, the surface of perovskite nanocrystals is susceptible to environmental changes. The Au nanocrystals formed through the reduction by oleylamine ligand are loosely bound to the semiconductor surface. These nonplasmonic (small size of $\sim 1\text{nm}$) nanoparticles accept electrons from the perovskite particles to become charged and (eventually) be expelled during steady state photoirradiation. Ultrafast measurements show that electron transfer to Au occurs from both hot and band edge electrons in the CsPbBr_3 conduction band, with rate constants of $(2.43 \pm 0.39) \times 10^{12} \text{ s}^{-1}$ and $(1.73 \pm 0.27) \times 10^{10} \text{ s}^{-1}$, respectively. The charged gold particles are then repelled from the negatively charged

CsPpbBr₃ surface, thus necessitating a stronger linkage between the two. The use of a bifunctional molecular bridge to establish a stronger linkage between the semiconductor and metal nanoparticles (e.g., mercaptopropionic acid employed to link TiO₂ and Au nanoparticles⁶⁴) is needed to develop a stable heterostructures.

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Supporting Information The supporting information include control experiments, histogram analysis, emission spectra and transient absorption measurements at different Au concentrations. The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/XXXX>

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