APPLIED PHYSICS

Relations between absorption, emission, and excited state chemical potentials from nanocrystal 2D spectra

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For quantum-confined nanomaterials, size dispersion causes a static broadening of spectra that has been difficult to measure and invalidates all-optical methods for determining the maximum photovoltage that an excited state can generate. Using femtosecond two-dimensional (2D) spectroscopy to separate size dispersion broadening of absorption and emission spectra allows a test of single-molecule generalized Einstein relations between such spectra for colloidal PbS quantum dots. We show that 2D spectra and these relations determine the thermodynamic standard chemical potential difference between the lowest excited and ground electronic states, which gives the maximum photovoltage. Further, we find that the static line broadening from many slightly different quantum dot structures allows single-molecule generalized Einstein relations to determine the average single-molecule linewidth from Stokes' frequency shift between ensemble absorption and emission spectra.

INTRODUCTION

Electronically excited molecules and materials can store energy (1, 2), generate power (3, 4), or emit light (1, 4). For homogeneous materials such as bulk crystalline semiconductors, the thermodynamic maximum photovoltage can be determined from the rate constants for photon absorption and emission as a function of frequency (4). Determination uses a generalization of Einstein's rate theory for absorption, stimulated emission, and spontaneous emission from gasphase line spectra (5) to thermalized, homogeneously broadened condensed-phase spectra (6-11). In condensed phases, thermalization shifts the emission spectrum to lower frequency than the absorption spectrum, this is known as Stokes' shift. The homogeneous generalized Einstein relations connect the equilibrium Stokes' shift to the linewidth. These also connect the rate constants for absorption and emission to the standard chemical potential that determines the maximum photovoltage. These relations are widely applicable to impurity ions in crystals (11) and bulk crystalline semiconductors (3, 4). In contrast, the relations have appeared to be inapplicable to molecular spectra (10); we suggest that distinct isomers (7, 12) or protonation states (7) in the ionic dye solutions used to test the relations may contribute to both this apparent inapplicability (10) and the apparent inhomogeneity of some dyes in photon echo experiments (12). The rate constant measurements needed to determine the standard chemical potential are not easy (4), and the approach is not valid if spectra are broadened by static inhomogeneities (10), such as the conformational heterogeneity of photosynthetic proteins or the size distribution of quantum dots.

A quantum dot is a semiconductor nanocrystal large enough to have bulk lattice properties but small enough that three-dimensional (3D) quantum confinement increases its bandgap above the bulk bandgap (13). Nanocrystal applications benefit from narrow size

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and structure distributions (14). With particle-in-a-box quantum confinement, the size distribution from a synthesis generates a static bandgap distribution that is diagnostic for size dispersion. For polydisperse samples, the resulting broadening of the absorption spectrum has successfully guided synthetic efforts to reduce the size dispersion. But even with narrow size distributions a few atomic layers wide (15-18), colloidal nanocrystals have an astonishing variety of atomic structures (14), emission spectra of single quantum dots differ from each other (18), and there is no homogeneous spectrum that is the same for every quantum dot. Instead, a quantum dot sample has a static inhomogeneous distribution of bandgaps and an average single-dot lineshape for each bandgap called the dynamical lineshape (19). The dynamical lineshape arises from the motions of electrons, holes, spins, ligands, and vibrations of the atomic lattice. Every homogeneous lineshape can be correctly called a dynamical lineshape, but dynamical lineshapes are not always homogeneous. For historical reasons, dynamical lineshapes have often been called homogeneous even when single-molecule spectroscopy has shown that the sample does not have a homogeneous lineshape. Unexpectedly, nonlinear spectroscopy has shown that the dynamical emission linewidth can be modified by a capping shell (20). As a result, in contrast to previous assumption, the total static and dynamic linewidth is not a reliable gauge for comparing size dispersion between samples with narrow static size distributions.

This realization, together with the thousands of electron microscopy image analyses required to characterize a size distribution, has motivated use of nonlinear and single-molecule spectroscopies to characterize dynamical linewidths in quantum-confined nanomaterials (17-20). By comparison to size/shape distributions from electron microscopy, femtosecond 2D Fourier transform (2DFT) spectroscopy (Fig. 1) has been proven to accurately measure bandgap inhomogeneity in quantum dot samples (19). With size/shape deviations on the order of an atomic layer, the bandgap distribution is quite accurately Gaussian (19). Because absorption and emission both contribute to 2D spectra (21-23), 2D separation of static inhomogeneity provides a way to test relations between dynamical absorption and emission spectra and use dynamical spectra to measure the standard chemical potential of excited states in heterogeneous samples.

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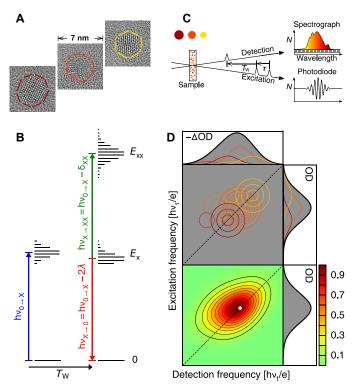


Fig. 1. The quantum dot size distribution diagonally elongates 2D spectra. (A) Transmission electron microscopy (TEM) images show the crystal lattices of three PbS dots from the ligand-covered ensemble. Maroon, orange, and yellow outline larger, average, and smaller dots, respectively. (B) Quantum dot energy bands (ground 0, exciton X, and bi-exciton XX) and optical transitions at thermal guasi-equilibrium. Exciton absorption has frequency $v_0 \rightarrow x$; after a sufficient waiting time T_{w} , exciton stimulated emission has frequency $v_{X \to 0} = v_{0 \to X} - 2\lambda/h$ and exciton to bi-exciton absorption has frequency $v_{X \to XX} = v_{0 \to X} - \delta_{XX}/h$. (C) The 2D pulse sequence uses two collinear pump pulses separated by an excitation delay τ , followed, after T_{w} , by a noncollinear probe pulse. The change in probe spectrum is detected by a spectrograph. (**D**) At fixed T_{w} , 2D spectra are obtained by Fourier transforming detected probe spectra with respect to excitation delay τ and displayed as contour maps of optical density change as a function of excitation and detection frequency. 2D signals from three dot sizes are colored to match TEM image outlines (top). Smaller dots are excited and detected at higher frequencies, diagonally elongating the ensemble 2D spectrum (bottom).

Here, we formulate single-molecule generalized Einstein relations and present an equation for the 2D spectrum in terms of singlemolecule Einstein *B* coefficient spectra. We record high-accuracy macroscopic 2D spectra and transform them into these Einstein *B* coefficient 2D spectra by removing pulse and propagation effects. We test the generalized Einstein relations on PbS quantum dot ensembles from two different synthetic procedures by developing least squares fit algorithms for 2D spectra. Dynamical absorption and emission spectra from such 2D fits provide the result we sought, a determination of the standard chemical potential change.

We were surprised to find a second remarkable result from these 2D fits. Quantum dot heterogeneity gives an ensemble-average Stokes' shift equal to the Stokes' shift for quantum dots with the average bandgap. This Stokes' shift and the single-molecule generalized Einstein relations can determine the dynamical linewidth. Together, these two results prove that the dynamical linewidth and static heterogeneity can be determined from ensemble absorption and emission spectra. We validate this determination method using literature data from fluorescence correlation spectroscopy (17).

RESULTS

Single-molecule generalized Einstein relations

The Einstein coefficients are single-molecule rate coefficients that quantify strengths for transitions between two quantum mechanical levels (5, 24). Einstein implicitly assumed that degenerate quantum states within each level equilibrate rapidly. For broadened spectra, the frequency spectra of the Einstein coefficients are related after the excited band reaches internal thermal quasi-equilibrium, where equilibrium holds within each band but not necessarily between bands. For example, within a few picoseconds after absorption of light, quasi-Fermi levels are established for each electronic band in bulk semiconductors (4), and electronically excited molecules vibrationally thermalize in condensed phases (25).

We begin by defining rate coefficients for three elementary rate processes with broadband spectra. The probability per unit time that a molecule *j* in the ground band 0 will absorb a photon and make a transition to the excited band X is

$$\Gamma_{j,0\to X} = \int_0^\infty b_{j,0\to X}(\nu, p, T) u(\nu) \, d\nu \tag{1A}$$

where $b_{j,0 \rightarrow X}(v, p, T)$ is the Einstein absorption *B* coefficient spectrum (absorption-*b*), u(v) is the electromagnetic energy density per unit frequency v, p is the pressure, and T is the absolute temperature. The probability per unit time that radiation will stimulate a molecule *j* in the excited band X to emit a photon into the same radiation mode and make a transition to the ground band 0 is

$$\Gamma_{j,\mathbf{X}\to0}^{b} = \int_{0}^{\infty} b_{j,\mathbf{X}\to0}(\mathbf{v},p,T) u(\mathbf{v}) d\mathbf{v}$$
(1B)

where $b_{j,X \to 0}(v, p, T)$ is the Einstein stimulated emission *B* coefficient spectrum (emission-*b*). Last, in the absence of radiation, the probability per unit time that a molecule *j* in the excited band X will spontaneously emit a photon and make a transition to the ground band 0 is

$$\Gamma_{j,\mathrm{X}\to0}^{a} = \int_{0}^{\infty} a_{j,\mathrm{X}\to0}(\mathsf{v},p,T) \, d\mathsf{v} \tag{1C}$$

where $a_{i, X \to 0}(v, p, T)$ is the Einstein spontaneous emission *A* coefficient spectrum (emission-*a*). The total probability per unit time that a molecule *j* in the excited band X will emit a photon and make a transition to the ground band 0 is $\Gamma_{j,X\to0} = \Gamma_{j,X\to0}^a + \Gamma_{j,X\to0}^b$.

With the assumptions of Boltzmann statistics, thermal quasiequilibrium within each band, and a constant refractive index n, the generalized Einstein relations connect absorption and emission transitions between the same two bands

$$a_{j,X\to 0}(v) = \frac{8\pi h v^3 n^3}{c^3} b_{j,X\to 0}(v)$$
(2A)

$$b_{j,X\to 0}(v,p,T) = b_{j,0\to X}(v,p,T) \exp\left[-(hv - \Delta\mu_{j,0\to X}^{0}(p,T))/kT\right]$$
(2B)

where *h* is Planck's constant, *c* is the speed of light in vacuum, *k* is Boltzmann's constant, and $\Delta \mu_{j,0\to X}^{o}$ is the difference in thermodynamic standard chemical potential between the excited band X and the ground band 0 for molecule *j* (9). $\Delta \mu_{j,0\to X}^{o}$ is an intrinsic property of a single molecule and its environment, specified by solvent, pressure, and temperature; (26) it is independent of the standard

states. The first relation is derivable from quantum electrodynamics (27) by assuming kinetic rates for spontaneous and stimulated emission in a transparent and nondispersive medium. The second relation is derivable from the first by applying detailed balance [equal forward (absorption) and reverse (total emission) rates $N_{j,0}^{eq} \Gamma_{j,0\to X} = N_{j,X}^{eq} \Gamma_{j,X\to 0}$] to single-molecule kinetics at thermal equilibrium. Equilibrium requires both a Planck blackbody radiation spectrum inside the refractive medium [equation 301 of (28)] and a Boltzmann population ratio (26) between bands, $(N_{j,X}^{eq}/N_{j,0}^{eq}) = \exp(-\Delta\mu_{j,0\to X}^{o}(p,T)/kT)$. The vacuum Einstein relations are recovered with delta function lines $b_{X\to 0}(v) = B_{X\to 0}^{v}\delta(v - v_{X\to 0})$ and $\Delta\mu_{0\to X}^{o} = hv_{X\to 0} - kT \ln(g_X/g_0)$, where g_X is the degeneracy of level X. Each Einstein coefficient is equal to the frequency integral of its spectrum.

Although the relationship between absorption and emission rate coefficients is obtained from a consideration of thermal equilibrium, it is valid whenever thermal quasi-equilibrium holds within each band. Lifetime broadening precludes thermal quasi-equilibrium, so Eq. 2B does not apply to lifetime broadened Lorentzian lineshapes in atoms. It can be violated if emission originates from two or more partially equilibrated bands (for example, excitons and defects) but the violation is diagnostically useful. The frequency-dependent exponential in Eq. 2B predicts that the quasi-equilibrium stimulated emission peak will always be shifted to a lower frequency than the absorption peak and that this shift becomes evident for linewidths comparable to the thermal energy kT (the Supplementary Materials illustrate this for Gaussian lineshapes); this frequency difference between peak maxima for *b*-spectra is Stokes' shift.

The difference in standard chemical potential $\Delta \mu_{0\to X}^{o}(p, T)$ is the difference in standard Gibbs free energy per molecule; $\Delta G_{0\to X}^{o}(p, T) = N_A \Delta \mu_{0\to X}^{o}(p, T)$ where N_A is Avogadro's number. In reactions that produce fuels, such as photosynthesis (1, 2), it gives the driving force for the chemical equilibrium constant. At constant pressure and temperature,

$$\Delta \mu = \Delta u + p \Delta v - T \Delta s \tag{3}$$

where Δu is the change in molecular internal energy, Δv is the change in molecular volume, and Δs is the change in molecular entropy (26). For photo-excitation, the pressure-volume term is usually negligible [based on the pressure dependence of PbS quantum dot absorption (29), changes in lattice spacing and compressibility give a shift of $\approx 10^{-6}$ eV at atmospheric pressure]. Standard molecular entropy increases arise from an increased number of electron/hole/spin configurations and their intraband splittings, from changes in lattice and ligand vibrations, from changes in nanocrystal volume, and from the solvent. The difference in chemical potential depends on the difference in standard chemical potential and on the population ratio (26)

$$\Delta \mu_{0 \to X} = \Delta \mu_{0 \to X}^{0}(p, T) + kT \ln(N_X/N_0)$$
(4)

In solid state physics, $\Delta \mu_0 \rightarrow X$ is the light-dependent difference between quasi-Fermi levels in the conduction and valence bands (4, 30). In photovoltaics, sustained nonequilibrium population of the excited state can generate a voltage $V = \Delta \mu_0 \rightarrow X/e$, where *e* is the elementary charge. At thermal equilibrium, $\Delta \mu_0 \rightarrow X = 0$ and the voltage is zero; the voltage that can be generated under saturating light ($N_X \sim N_0$) is given by the difference in standard chemical potential.

Effects of a bandgap distribution on 1D and 2D spectra

The absorption and stimulated emission cross sections are proportional to the ensemble average of the corresponding *B* coefficient spectrum

$$\sigma_{0 \to X}(\mathbf{v}, p, T) = h \mathbf{v} \langle b_{j,0 \to X}(\mathbf{v}, p, T) \rangle_j n/c$$
(5)

The ensemble average denoted by angle brackets is an average over quantum dots *j* with different sizes and shapes, which gives rise to static inhomogeneous broadening of the absorption and emission spectra. Equation 2B is valid for single quantum dots but does not hold for ensemble-averaged quantum dot spectra because quantum dots of different size have different chemical potentials. The previously known relation (8-11) between macroscopic stimulated emission and absorption cross sections with the same form as Eq. 2B holds only for homogeneous samples. The Beer's law absorbance or optical density spectrum for an isolated electronic transition is proportional to the absorption cross section

$$OD(\mathbf{v}, p, T) = N_0 \left[\sigma_{0 \to X}(\mathbf{v}, p, T) / \ln(10) \right] \ell$$
(6)

where OD is the optical density, N_0 is the molecular number density in band 0, and ℓ is the sample path length. The transformation of absorbance and photoluminescence (PL) spectra into absorption and emission $\langle b \rangle$ is shown below in Materials and Methods.

The absorptive Einstein *B* coefficient 2D spectrum of the ensemble is an average of single-molecule 2D spectra (Fig. 1D), each of which is a 2D product of the excitation probability spectrum with the sum of transition probability spectra for the three detected transitions in Fig. 1B

$$S_{2D}^{B}(\mathbf{v}_{t},\mathbf{v}_{\tau};T_{W}) = N_{X}(T_{W})$$

$$\langle b_{j,0\to X}(\mathbf{v}_{\tau}) [b_{j,0\to X}(\mathbf{v}_{t}) + b_{j,X\to0}(\mathbf{v}_{t}) - b_{j,X\to XX}(\mathbf{v}_{t})] \rangle_{j}$$
(7)

where the pressure and temperature dependence has been suppressed and $N_{\rm X}$ is the number density of the excited band X, which depends on the waiting time $T_{\rm w}$. v_{τ} is the excitation frequency and v_t is the detection frequency. In Eq. 7, $b_{j,0\to X}(v_\tau)$ is the absorption probability for the excitation step, which depletes the ground band 0 and populates the single-exciton band X. Excitation changes the optical density of the sample in three ways: depleting 0 causes a reduction in $0 \rightarrow X$ absorption [more detected photons, positive signal $b_{i,0\to X}(v_t)$; populating X causes both X $\rightarrow 0$ single exciton stimulated emission [more detected photons, positive signal $b_{i, X \rightarrow 0}(v_t)$] and $X \rightarrow XX$ exciton to bi-exciton absorption [fewer detected photons, negative signal $b_{i, X \to XX}(v_t)$] (23). 2D spectra from each single-molecule add together to give the ensemble 2D spectrum. Fig. 1 shows how 2D spectra can separate the dynamical broadening perpendicular to the diagonal from the total broadening parallel to the diagonal, allowing a test of the generalized Einstein relation between B coefficient spectra. Furthermore, a single 2D spectrum also puts the dynamical absorption and emission b on the same intensity scale, allowing determination of the change in standard chemical potential upon excitation.

High-accuracy 2D spectra show stimulated emission and enable quantitative fitting

For 2D experiments, colloidal PbS quantum dots with a bandgap of 1.07 eV were prepared by two different air and moisture free syntheses: one sample was synthesized from lead chloride and sulfur in

oleylamine (15) and capped with 3.1 oleate ligands/nm² (covered sample); the second sample was synthesized from lead oleate and substituted thiourea (16) and capped with 1.1 oleate ligands/nm² (part-covered sample). Figure S1 shows electron microscopy images and histograms that reveal the size and shape distribution (table S1). The transformation of experimental absorbance and PL spectra into Einstein $\langle b \rangle$ transition probability lineshapes for absorption and stimulated emission is illustrated in Materials and Methods. Thermally equilibrated 2D spectra are measured for air and moisturefree PbS quantum dots in a sealed, interferometrically stable spinning sample cell (31) that rotates to a fresh sample for every laser shot, thus eliminating contamination of the 2D spectra by photocharging (32). Without the usual measures to reduce absorption or red-shift emission (33), this enables the observation of singleexciton stimulated emission needed to test the generalized Einstein relations. For high accuracy, the 2D spectra are recorded using a set of excitation pulse pair delays obtained by step-scanning a Mach-Zehnder interferometer (fig. S2) that is actively stabilized to ±0.6 nm pathlength difference at each step. As recorded, each 2D spectrum is attenuated by the pulse spectra and sample absorption. Materials and Methods shows the transformation to an Einstein B coefficient 2D spectrum through division by pulse spectra and propagation functions.

Figure 2A shows an experimental Einstein B coefficient 2D spectrum of PbS quantum dots with partial ligand coverage at T_w = 1 ps waiting time, after intraband relaxation (34, 35). The most notable aspects of the experimental 2D spectrum in Fig. 2A are the strong stimulated emission around the white cross and the absence of the usual region with net negative signal arising from absorption by the excited state (19). These features were revealed by air-free synthesis and elimination of repetitive excitation with the spinning sample cell (31). Since stimulated emission by itself can only shift the 2D maximum to lower detection frequency, as in Fig. 1D, the slight shift of the 2D maximum to higher detection frequency (see the contours around the white dot) indicates that exciton to bi-exciton absorption lies only slightly to lower detection frequency than the depleted ground state absorption. Partial cancelation between these two overlapping and oppositely signed signals makes stimulated emission appear stronger.

An algorithm was developed to least-squares fit the 2D spectra to a model containing the three signal contributions in Fig. 1B and Eq. 7. With size/shape deviations on the order of an atomic layer on each face, the bandgap distribution is quite accurately Gaussian, as shown by transmission electron microscopy and 2D spectroscopy (19). Starting from the ensemble absorption and emission, the algorithm (flowchart in fig. S3) first finds dynamical absorption and emission by deconvolution with the static bandgap distribution [a Gaussian with standard deviation (SD) Δ_{static}]. The next step shifts (δ_{XX}) and broadens (by convolution with a Gaussian of SD Δ_{XX}) the dynamical absorption to get the dynamical exciton to bi-exciton absorption. We use eq. S5 to sum these three terms with variable strengths for emission $(w_{X \to 0})$ and exciton to bi-exciton absorption $(w_{X \to XX})$ and use eq. S9 as an algorithmically convenient form for the 2D spectrum in Eq. 7. The product 2D spectrum in Eq. 7 is then ensembleaveraged over the bandgap distribution, and all five parameters are optimized. This fit always matches the ensemble absorption and stimulated emission spectra exactly-only the 2D spectrum is treated as data to be fit. The fit (Fig. 2, B and C) has a reduced chisquared of 1.04. All five parameters are necessary (fig. S4), and the

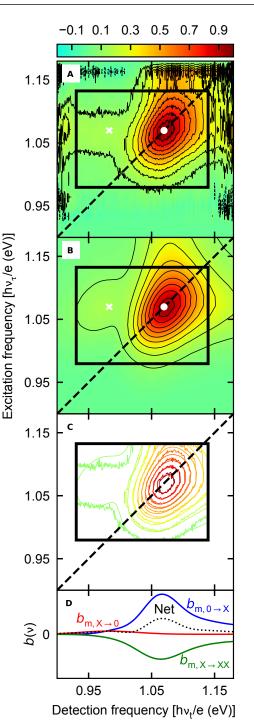


Fig. 2. Comparison between experimental and fit Einstein *B* coefficient 2D spectra for the partially ligand-covered PbS quantum dot sample. The color bar at the top is used to show signal as a fraction of maximum signal. (A) Experimental 2D spectra with black contours at 10 to 90% of maximum. The rectangle encloses the fitted region. The dashed line shows the diagonal, the white dot marks the absorption- $\langle b \rangle$ maximum, and the white cross marks the emission- $\langle b \rangle$ maximum. (B) Least squares fit to the 2D spectrum using eq. S9. (C) Overlaid 10 to 90% contours of the experimental (2.4% noise) and fit (smooth) 2D spectra. (D) The net signal (black dotted curve) and three dynamical contributions for the most probable quantum dot size and bandgap (*m*): positive reduced exciton absorption (blue curve), positive exciton stimulated emission (red curve), and negative exciton to bi-exciton absorption (green curve).

Table 1. Best fit parameters for 2D spectra of the two PbS quantum dot samples. Asymmetrical error bars are given in parentheses for each fit parameter and result. The change in standard chemical potential is given for quantum dots with the most probable bandgap, $v_{m,0\rightarrow X} = v_{0\rightarrow X}^{ens}$. For reference, bi-exciton broadening increases the exciton to bi-exciton lineshape FWHM by 12.44 meV in the best fit for the covered sample at $T_w = 1$ ps. The half-width at half maximum (HWHM) is used to quantify the dynamical linewidth for the 1S-1S transition because its high frequency side is overlapped by transitions to higher energy states.

Fit parameters	Part-covered, $T_w = 1 \text{ ps}$	Covered, $T_w = 1 \text{ ps}$	Part-covered, $T_w = 5 \text{ ps}$
Δ_{static}/e (meV)	23.9 (-0.5,+0.5)	27.4 (-0.7, +0.7)	23.8 (-0.5, +0.5)
δ _{XX} / <i>e</i> (meV)	7.4 (–2.8, +3.7)	4.8 (-3.5, +9.3)	5.8 (-2.6, +3.5)
Δ _{XX} / <i>e</i> (meV)	19.7 (-3.4, +3.2)	16.3 (–7.3, +7.6)	15.6 (–2.7, +2.9)
W _{X→0}	6.9 (-1.9, +2.2)	7.2 (-4.4, +6.5)	5.2 (–2.0, +2.1)
$W_{X \rightarrow XX}$	49.9 (-2.8, +8.8)	49.4 (–9.6, +20.8)	52.5 (-5.0, +9.0)
Fit results			
χ ²	2901	6256	7864
χ^2_{v}	1.04	1.99	2.83
HWHM _{dyn} /e(meV)	36.0 (-0.4, +0.4)	42.6 (-0.6, +0.5)	36.4 (-0.6, +0.7)
Δμ _{<i>m</i>,0→χ} / <i>e</i> (eV)	0.97 (-0.01, +0.01)	0.97 (-0.03, +0.02)	0.97 (-0.02, +0.01)
hν ^{ens} _{0→X} /e (eV)	1.070	1.069	1.070
$h v_{X \to 0}^{\text{ens}} / e$ (eV)	0.982	0.985	0.982

2D width (fig. S5) provides evidence for bi-exciton broadening (Fig. 2D). Table 1 has parameters and error bars for 2D spectra, including a later waiting time and the covered sample (fig. S6).

The ensemble linewidths are dominated by dynamical broadening. It has been reported that passivating ZnS shells can reduce the dynamical linewidths for CdSe quantum dots by up to one-third (20). We find that differences in shape and ligand coverage between samples change the dynamical broadening by about one-sixth of the dynamical linewidth, increasing the dynamical half width at half maximum from 36 meV (part-covered) to 43 meV (fully covered). The two samples naturally differ in static size dispersion broadening, with $\Delta_{\text{static}}/e = 24$ versus 27 meV. All remaining parameters agree within error for both samples, which is consistent with intrinsic properties of PbS quantum dots.

Where comparison is possible, agreement with prior work is good. The shift $\delta_{XX}/e = 7$ meV of the exciton to bi-exciton transition agrees with a prior measurement using two-photon excitation (36). In bulk PbS, the valence and conduction bands both have an eightfold degeneracy when electronic momentum, angular momentum, and spin are included. The ground state, with a full valence band and empty conduction band, has only one electron-spin configuration. The exciton, with one hole in the valence band and one electron in the conduction band, has $64 = 8 \times 8$ electron-hole-spin configurations (37, 38). According to a widely used degeneracy model (39) with electron-hole-spin configuration splittings that are small relative to the thermal energy k_BT , the relative strengths of depleted ground state absorption, exciton to bi-exciton absorption, and stimulated emission should be $w_{0 \rightarrow X}$: $w_{X \rightarrow XX}$: $w_{X \rightarrow 0} = 64:49:1$ in PbS quantum dots. The ground to exciton and exciton to bi-exciton absorption strengths match the predicted 64:49 ratio.

Table 1 also presents internally consistent new results. Since the Stokes' shift generally arises from a combination of electron-hole-spin configuration splittings and exciton-phonon interactions (40),

the degeneracy model must implicitly attribute the Stokes' shift of $\sim 4k_BT$ to exciton-phonon coupling. However, stimulated emission is about seven times stronger than predicted by the degeneracy model. This stronger emission indicates that relaxation down to energy levels that are lowered by electron-hole-spin configuration splitting and bright in emission plays an important role in red-shifting the emission by more than the thermal energy. The splittings are a property of the exciton and may depend on the atomic structure of an individual quantum dot (41). On the basis of the above explanation of the excess emission strength from electron-hole-spin configuration splittings, we expect broadening in PbS quantum dots because of the increase from 64 electron-hole-spin configurations in the exciton band to 784 in the bi-exciton band. Such broadening of the exciton to bi-exciton transition is known in quantum wells (42). The best fit bi-exciton broadening is ($\Delta_{XX}/e \sim 18$ meV). The bi-exciton broadening is expected to be approximately the same as the broadening of the exciton transition from electron-hole-spin configuration splittings, so the overall dynamical linewidth is consistent with previously reported exciton-phonon couplings for vibrations of the nanocrystal lattice (35). Since the high-accuracy 2D spectra do not show a bare negative signal from the exciton to bi-exciton transition, these remarkably consistent bi-exciton parameters are a key result of the least squares fit algorithm developed here. The dynamical absorption and emission spectra are found to agree with the generalized Einstein relations; each predicts the maximum for the other within 5 meV. At the most probable bandgap, the change in standard chemical potential is $\Delta \mu_{m,0 \to X}^{0}/e = 0.97 \pm 0.01 \text{eV}$ for both samples.

Tests of the generalized Einstein relations

A second fit of the 2D spectra (flowchart in fig. S7) enforced the generalized Einstein relations as a constraint. In this second fit, the dynamical absorption and stimulated emission spectra are related

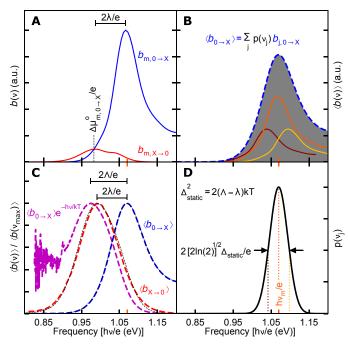


Fig. 3. Einstein B coefficient spectra of part-covered PbS quantum dots obtained by simultaneously fitting absorption, emission, and 2D spectra with a Gaussian static bandgap distribution while enforcing the single-molecule generalized **Einstein relations.** (A) The dynamical absorption-*b* (blue curve) and stimulated emission-b (red curve) are plotted on the same scale for the most probable bandgap hvm (orange tickmarks). Dynamical absorption and emission become equal at the most probable change in standard chemical potential ($\Delta\mu^o_{m,0\to\chi})$ between ground and exciton bands. (B) The ensemble-averaged absorption spectrum (blue dashed curve) and three probability-weighted dynamical absorption spectra (maroon, orange, and yellow curves) in the average. (C) The ensemble-averaged absorption (blue dashed curve) and generalized Einstein relations can test for static broadening by calculating a hypothetical homogeneous stimulated emission spectrum (purple dashed curve) that overestimates the Stokes' shift (2 Λ). Comparison to the ensemble-averaged stimulated emission spectrum (red dashed curve) and Stokes' shift (2λ) indicates static broadening. The black dotted curves show the ensembleaveraged absorption and stimulated emission spectra calculated from the dynamical spectra, which have Stokes' shift (2 λ) marked above. (**D**) The Gaussian probability distribution of static bandgaps is centered on the most probable bandgap hv_m with variance $\Delta_{\text{static}}^2 = 2(\Lambda - \lambda) kT$. Dotted vertical lines mark static bandgaps for the three dynamical absorption spectra in (B). a.u., arbitrary units.

by Eq. 2B, and the ensemble absorption lineshape, ensemble stimulated emission lineshape, and 2D spectrum are all data to be fit. The results are shown in Fig. 3A for the center of the static distribution. The dynamical emission lineshape is quite asymmetric, with a strong high-energy shoulder. A similar asymmetry has been reported in single-molecule PbS/CdS core/shell quantum dot emission spectra and attributed to intraband splittings (*38*). Calculations (*37*) have predicted a corresponding asymmetry in absorption. The 2D spectra reveal a weak low-energy shoulder in dynamical absorption that corresponds to the dynamical emission maximum and shows that the high-energy dynamical emission shoulder corresponds to the dynamical absorption maximum.

The change in standard chemical potential, $\Delta \mu_{0\to X}^{o}$, can be found from Eq. 2B as the photon energy at which dynamical absorption and emission have equal *b*. This determination in Fig. 3A is possible because 2D spectroscopy puts the dynamical absorption and emission

spectra on a common intensity scale (eq. S10 gives the connection to the weights in Table 1). $\Delta \mu_{m,0\to X}^{o} = 0.98 \text{ eV}$ from Fig. 3A is coincidentally equal to the dynamical emission maximum and agrees with the unconstrained fit result within error. As a point of reference, using a Gaussian linewidth with SD Δ_{vib} = 25 meV for the lattice vibrations or phonons and the three bright levels from the calculation in (38) to calculate the absorption and emission spectra of the bright exciton would give a lower emission weight of $w_{X\to 0} \approx 5$ and put the standard chemical potential change about 20 meV below the emission maximum. Given the perfectly faceted quantum dots in (38), this is remarkable agreement. Compared to the photon energy at the absorption maximum, the standard chemical potential change for this three bright level model is lowered about 12 meV by lattice vibrations and ligand motions (an internal energy effect) and about 80 meV by intraband relaxation (divided almost evenly between energetic and entropic effects) among the energetically split levels of the bright exciton.

The change in standard chemical potential also affects the spontaneous emission rate, as can be seen from Eqs. 1C and 2. Using the emission weight from the 2D spectra, the calculated radiative lifetime (see Materials and Methods) for the bright exciton is about 200 ns, consistent with an initial fast PL decay in figure 3D of (17). Without any changes in the emission spectrum, the spontaneous emission rate becomes slower as the bright exciton equilibrates with the nonradiating dark exciton. After 1 µs, the PL decay has a single-exponential lifetime of about 2 µs (17). This 2-µs lifetime matches the radiative lifetime calculated for thermal quasi-equilibrium among all 64 exciton states (the levels from (38) give an emission weight $w_{X\to 0} \approx 0.7$). This agreement indicates a parallel with the long radiative lifetime in PbSe quantum dots, which arises from quasi-equilibrium with dark states (37). Slowing of the radiative rate by a factor of 10 corresponds to lowering the standard chemical potential by about 50 meV as the bright exciton equilibrates with the dark exciton. In other words, the standard chemical potential of the quasi-equilibrated exciton is 50 meV lower than the standard chemical potential of the bright exciton measured here.

Using ensemble absorption and emission to determine inhomogeneity

The single-molecule generalized Einstein relations can be directly applied to ensemble spectra as a test for static inhomogeneity. Figure 3B illustrates ensemble-averaging with the bandgap distribution in Fig. 3D. Figure 3C compares the ensemble absorption and emission $\langle b \rangle$ from this constrained fit to experiment. Figure 3C also shows the ensemble emission lineshape predicted under the hypothesis of thermal quasi-equilibrium and homogeneity using Eq. 2B: $\langle b_{X\to 0}(v) \rangle \propto \langle b_{0\to X}(v) \rangle \exp(-hv/kT)$. For this hypothesis test, zero baseline levels are needed to avoid exponential amplification of background or noise over the signal. If the absorption and emission lineshapes are mutually incompatible with Eq. 2B, the incompatibility is diagnostic of partially equilibrated emission (for example, exciton emission faster than quasi-equilibrium with a defect or trap). On the other hand, if the lineshapes are compatible except for their Stokes' shift, comparison between the hypothetical Stokes' shift calculated from the ensemble spectra (2Λ) and the experimental Stokes' shift (2λ) can quantify the bandgap distribution of the ensemble. Figure 3C compares this hypothetical homogeneous emission lineshape to experiment. Since the ensemble absorption is broader than dynamical absorption, this test overestimates the ensemble Stokes' shift as 2Λ . As can be seen by comparing Fig. 3 (A and C), the observed ensemble Stokes' shift matches the dynamical Stokes' shift for the center of the bandgap distribution and is not appreciably altered by the static inhomogeneity. This overestimate of the Stokes' shift provides an easily measurable, model-free signature of static inhomogeneity in an ensemble.

Simple quantitative relationships are specific to the common case of Gaussian lineshapes. For a Gaussian dynamical absorption lineshape with variance Δ_{dyn}^2 (so that Δ_{dyn} is its SD), the dynamical stimulated emission will have a Gaussian lineshape with the same width and a Stokes' shift of 2λ that obey

$$\Delta_{\rm dyn}^2 = 2\lambda kT \tag{8}$$

Equation 8 is a consequence of the generalized Einstein relations for any Gaussian dynamical lineshape and says that larger dynamical linewidths necessarily arise from larger Stokes' shifts. Neglecting variation in the Stokes' shift with bandgap, convolution with a Gaussian static lineshape with variance Δ^2_{static} gives Gaussian ensemble absorption and emission lineshapes with the same increased variance

$$\Delta_{\rm ens}^2 = \Delta_{\rm dyn}^2 + \Delta_{\rm static}^2 \tag{9}$$

Because the high-frequency side of the lowest exciton absorption typically has an asymmetrical shape caused by transitions to higher exciton states, the ensemble full width at half maximum (FWHM) for emission gives the more reliable $\Delta_{ens} = FWHM_{ens}/[2\sqrt{2ln(2)}]$. The experimental Stokes' shift of 2λ can be used to calculate the dynamical lineshape variance Δ_{dyn}^2 from Eq. 8. Together, the ensemble and dynamical variances determine the static line broadening using Eq. 9.

Alternatively, the static line broadening can be determined from

$$\Delta_{\text{static}}^2 = 2(\Lambda - \lambda) kT \tag{10}$$

as illustrated in Fig. 3D, which relies on Gaussian convolution in a different way. Eqs. 8 to 10 independently determine all three linewidths and can be combined to act as a consistency check for Gaussian line broadening. The variance, FWHM and Stokes' shift also increase or decrease together for other lineshapes. Practically, Eq. 9 is useful for ensemble absorption and emission lineshapes that approximate Gaussians with the same linewidth in between their maxima, while Eq. 10 requires that 2A be calculated by applying Eq. 2B to a high signal to noise lineshape. For the PbS quantum dots studied here, the latter route determines $(2\Lambda)/e = 112$ meV, so that Eq. 10 gives $\Delta_{\rm static}/e = 24$ meV, which agrees with the 2D estimate within error.

We have tested these relationships using Gaussian dynamical emission spectra from the literature. The test for photon-correlation Fourier spectroscopy of PbS quantum dots (17) is shown in fig. S8, and tables S2 and S3 list the data points plotted in fig. S8. For the four smaller PbS quantum dot sizes with dynamically broadened emission spectra (17), the ensemble Stokes' shift and ensemble emission linewidth exemplify the generalized Einstein relation result for homogeneous Gaussian lineshapes in Eq. 8. The generalized Einstein relations also reveal that four ensembles of larger size PbS quantum dots, for which photon correlation Fourier spectra were not reported in (17), have previously unrecognized static size distribution broadening.

DISCUSSION

Outlook for using the Stokes' shift to quantify static and dynamic linewidths

For the two different PbS quantum dot samples with the same bandgap studied here, the dynamical linewidths account for over 90% ($\Delta_{dyn}/\Delta_{ens}$) of the ensemble emission linewidths. However, the dynamical linewidths differ by more than 15% between the ligand-covered and part-covered samples. This comparison shows that ensemble linewidths cannot be used to assess size distributions with a width of a few atomic layers, so new methods are needed.

The single-molecule generalized Einstein relations between absorption and thermalized emission spectra of the same transition relate the Stokes' shift between Einstein *b* lineshapes to the dynamical linewidth. Use of the ensemble-averaged Stokes' shift to determine the dynamical linewidth in heterogeneous semiconductor nanomaterials is likely to be widely practical for five reasons: First, bulk semiconductors typically attain the required thermal quasi-equilibrium within a few picoseconds; second, nanomaterial emission lifetimes are usually long enough that the measured time-integrated PL is dominated by emission at thermal quasi-equilibrium; third, quantumconfined nanomaterial Stokes' shifts are usually small enough that the overlap between absorption and emission spectra is dominated by the same transition so that the generalized Einstein relations apply; fourth, with these smaller Stokes' shifts, static ensemble broadening increases the absorption-emission overlap to permit numerically reliable cross-estimates of ensemble absorption and emission maxima from the generalized Einstein relation as a test of the homogeneity hypothesis; fifth, the semi-infinite variety of nanomaterial structures generates inhomogeneous distributions of the static bandgap that are nearly Gaussian for the same reasons the central limit theorem generates Gaussian distributions, simplifying the ensemble-dynamicalstatic lineshape relationship.

This conclusion does not contradict the well-known principle that a nonlinear spectroscopic measurement is required to differentiate homogeneous and inhomogeneous broadening. The required nonlinear spectroscopic measurement in the generalized Einstein relation/ Stokes' shift method presented here is measurement of the emission spectrum. PL is a two-photon process with the great advantage that it can be easily measured on commercial spectrofluorometers.

Outlook for excited state chemical potentials

The change in standard chemical potential for creation of a bright exciton is an intrinsic property of the quantum dot, independent of surface coverage to within 10-meV experimental uncertainty. Such independence is expected for the bulk (13) but seems remarkable for a 4-nm diameter quantum dot in which about one-third of the atoms lie on the surface. With picosecond thermalization, this chemical potential controls exciton dynamics. Figure 1B illustrates a non-equilibrium decrease in the internal energy of the bright exciton as it relaxes to thermal quasi-equilibrium during the waiting time.

Within a picosecond, the 2D spectra become stationary, indicating that energetic relaxation among bright levels is complete. On the same time scale, the decay of the cross-polarized transient grating signal (34, 35) indicates complete exciton-spin relaxation for the bright levels. Together, these two observations show that the bright exciton is at internal thermal quasi-equilibrium within a picosecond, but they are silent about the dark exciton. The standard chemical potential change (or the closely related stimulated emission strength) shows that the bright exciton has not yet appreciably equilibrated

with the dark exciton levels. Bright-dark exciton equilibration requires separate electron and hole angular momentum/spin relaxation. In CdSe quantum dots, electron-spin flips are reported to be over an order of magnitude slower than exciton-spin flips (43). It is not yet clear whether separate electron or hole total angular momentum relaxation in PbS quantum dots is slow enough to explain the few hundred nanosecond relaxation from the bright exciton to a more slowly emitting mixture of bright and dark excitons in thermal quasi-equilibrium with each other [in principle, defect (17) or trap states could also play a role]. However, relaxation between bright and dark excitons must be much slower than picoseconds to explain the stronger than expected single exciton emission in the 2D spectra, which arises from the bright exciton. During bright-dark equilibration, the 2D spectra would measure a nonequilibrium chemical potential; at longer waiting times, the 2D spectra would measure the standard chemical potential change for creating an exciton.

The standard chemical potential change for creation of an exciton generalizes electronic degeneracy factors to incorporate intraband splittings from quantum-confined electron-hole-spin interactions (even if greater than the thermal energy), changes in phonons, ligand, and solvent rearrangement. Although changes in quantum dot volume can affect all three terms in Eq. 3, these volume-change contributions cancel in the chemical potential (the change in chemical potential equals the maximum nonexpansion work). As can be seen from Eqs. 1C and 2, the change in standard chemical potential alters the spontaneous emission rate, which will alter thermalized (44) excitation energy transport.

2D spectroscopy allows measurement of the standard chemical potential for transient excited species present at a particular waiting time of interest. Thermalized excitation energy transport in quantum dot arrays can take place over tens of picoseconds (44), which is after the bright exciton has reached thermal quasi-equilibrium but before bright exciton relaxation toward equilibrium with the dark exciton. In these circumstances, the appropriate description of excitation energy transport uses the standard chemical potential for creation of a bright exciton.

Since the work of Förster on excited state proton transfer (45) and of Marcus on excited state charge transfer (46), photochemical equilibria have been treated using approximate estimates of standard internal energy changes from spectra. For understanding such photochemical equilibria, the necessary quantity has always been the standard free energy change, which is given by the standard chemical potential change. The use of 2D spectra to measure excited state thermodynamics through the standard chemical potential solves the previously known inhomogeneity roadblock to use of the generalized Einstein relations. It has also revealed a previously unknown problem for their use that arises from time-dependent nonequilibrium mixtures of metastable species. As seen here for PbS quantum dots, such nonequilibrium mixtures can have time-varying radiative rates that would frustrate attempts to measure their thermodynamics through spontaneous emission. 2D spectroscopy can solve this problem by measuring stimulated emission at specified waiting times. After quantum decoherence, nonequilibrium statistical mechanics and irreversible thermodynamics (47) govern reactions. 2D spectroscopic sensitivity to entropy opens up study of photochemical thermodynamics and relaxation in heterogeneous systems such as nanomaterials, polymers, glasses, and proteins. Just as femtosecond 2D spectroscopy enabled study of quantum coherent dynamics in heterogeneous systems (48-50), the Einstein B coefficient 2D spectra

and single-molecule generalized Einstein relations presented here enable study of the quasi-equilibrium thermodynamics that govern the efficiency of photovoltaics and light-emitting diodes in heterogeneous materials.

Limitations

The equations presented here do not include a wavelength-dependent refractive index or allow for appreciable absorption over distances on the order of one wavelength (as occurs in metals). A quantitative determination of static inhomogeneous broadening from 1D spectra using Eqs. 8 and 9 requires a single electronic transition with Gaussian lineshapes. This excludes poly-disperse samples in which the lowest exciton peak is only a shoulder. A smaller size dispersion that is still large enough to cause a systematic increase in the Stokes' shift with bandgap broadens ensemble absorption more than ensemble emission (the Supplementary Materials provide an estimate of this effect); in such cases, replacing convolution with a linear integral equation might improve accuracy. Because errors are amplified by the exponential, directly testing a homogeneity hypothesis with Eqs. 2B and 10 requires accurate background subtraction that yields a zero baseline for frequencies below the absorption band. This method can usefully predict an emission peak for samples in which the lowest exciton transition is only a shoulder in absorption. Eqs. 2B and 10 rely on absorbance and PL spectra that overlap each other with all emitting bands in mutual quasi-equilibrium. For example, partial equilibration between the exciton and a defect or trap state with overlapping emission can cause a diagnostically useful failure to capture one lineshape from the other, as can an inhomogeneous departure from proportionality between the ensembleaveraged spontaneous emission rate spectrum and the time-integrated PL spectrum (see the Supplementary Materials). Briefly, the 1D generalized Einstein relation/ensemble Stokes' shift method can determine dynamical and static line broadening for high-quality samples with small size dispersion, where simply comparing the width of the first exciton peak fails. 2D spectra can resolve inhomogeneities to recover asymmetrical dynamical lineshapes and static bandgap distributions so long as the waiting time exceeds the time required to attain thermal quasi-equilibrium. Determination of the standard chemical potential of the excited band requires a 2D spectrum without cumulative excitation effects to characterize single-exciton stimulated emission.

Conclusions

Using 2D spectra to test single-molecule generalized Einstein relations led to two methods for characterizing heterogeneous materials. Singlemolecule generalized Einstein relations between absorption and emission show that the Stokes' shift between ensemble absorption and emission spectra can quantify dynamical linewidths and static disorder in heterogeneous materials. Here, differences in shape and ligand coverage can alter the intrinsic dynamical linewidth by more than the static size dispersion alters the ensemble linewidths. As a result, the Stokes' shift estimates should replace use of the ensemble-averaged absorption linewidth for estimating size dispersion broadening. Further, the single-molecule generalized Einstein relations can be used to measure the chemical potential of excited bands by 2D spectroscopy. This measurement was made possible by using air-free quantum dot synthesis and shot-to-shot sample exchange, which led to strong stimulated emission in the 2D spectra. For determining the chemical potential, 2D experiments replace separate

absolute measurements of absorption and emission, which are restricted to homogeneous materials. 2D spectra probe both absorption and emission in a single measurement that resolves inhomogeneity and is thus applicable to disordered materials. The relative absorption and emission strengths in 2D spectra provide a contact-free determination of the standard chemical potential, which gives the maximum photovoltage a disordered material can generate.

MATERIALS AND METHODS

Materials

Low bandgap PbS nanocrystals were chosen so that a reported defect state (17) lies above the lowest exciton and is not involved in the experiment. The ligand-covered sample of lead sulfide (PbS) nanocrystals with lowest electronic absorption peak at 1.074 eV was synthesized by hot injection of elemental sulfur in oleylamine into lead (II) chloride in oleylamine (15) with a modified washing procedure (51). For literature comparison, the usual assumption of spherical nanocrystals gives a nominal oleate ligand surface coverage of ~3.1/nm². Chloride also passivates these nanocrystal surfaces. The partially ligand-covered sample of lead sulfide (PbS) nanocrystals with the lowest electronic absorbance peak at 1.072 eV was synthesized from lead oleate and a thiourea (16). Some oleate capping ligands were removed with 1,2-bis(dimethylphosphino)ethane in toluene solution for a nominal oleate surface coverage of $\sim 1.09/nm^2$. Annular dark field-scanning transmission electron microscopy images of both "covered" and "part-covered" samples show faceted nanocrystals (fig. S1). Table S1 lists similar average area equivalent circular diameters of 4.29(11) and 4.27(9) nm, respectively. The covered sample has a smaller major/minor axis projection ratio (~1.14) than the part-covered sample (~1.26). Optical experiments were performed on clear solutions of colloidal nanocrystals in tetrachloroethylene.

Methods

Absorbance spectra and conversion to Einstein absorption B spectra

Linear absorbance spectra (52) of covered and part-covered PbS nanocrystals were recorded in both the spinning sample cell used for 2D spectroscopy (0.27-mm pathlength) and the cuvette (10-mm pathlength) used for PL spectroscopy. The 400 to 1500 nm wavelength range was scanned with a 1-nm wavelength sampling interval. Accurate calculation of hypothetically homogeneous emission from absorbance requires that the ensemble absorption lineshape decay to zero baseline at frequencies below the peak. Figure 4 shows the transformation from absorption and PL spectra as a function of wavelength to absorption and stimulated emission Einstein *B* coefficient lineshapes for the part-covered sample.

The net absorption of light by a macroscopic sample is usually quantified with the decadic absorbance spectrum or optical density spectrum

$$I_{\ell}(\lambda) = I_0(\lambda) \, 10^{-\text{OD}(\lambda)} \tag{11}$$

where $I_0(\lambda)$ is the incident spectral irradiance, $I_{\ell}(\lambda)$ is the transmitted spectral irradiance, and λ is the vacuum wavelength [not to be confused with the Stokes' shift (2 λ) in Eqs. 8 and 10. Irradiance is sometimes called intensity. The optical density wavelength spectrum

is calculated as $OD(\lambda) = -\log_{10}[I_{\ell}(\lambda)/I_0(\lambda)]$. The spectral irradiance conversion from vacuum wavelength to frequency is

$$I(v) = -I(\lambda) d\lambda/dv = I(\lambda = c/v) c/v^2$$
(12)

The negative sign in the first equality arises from the reversal of axis direction between wavelength and frequency. For calculating the optical density frequency spectrum, the change of variables factor $d\lambda/dv$ common to the numerator and denominator in the irradiance ratio cancels, so that the optical density frequency spectrum is

$$OD(v) = OD(\lambda = c/v)$$
(13)

Slight changes in shape between Fig. 4 (A and B) arise solely from nonlinear distortion of the horizontal axis in the conversion from wavelength to frequency, $v = c/\lambda$. The quantum dot solutions are optically clear and assumed not to scatter light, so that attenuation arises only from absorption. In the frequency range with only ground to exciton absorption, Eqs. 5 and 6 give

$$\langle b_{i,0\to\mathbf{X}}(\mathbf{v})\rangle_i = [\ln(10)\operatorname{OD}(\mathbf{v})/(N_0\ell)](c/nh\mathbf{v})$$
(14)

where N_0 is the number density of molecules in the ground band 0 (units: molecules/volume), ℓ is the sample cell pathlength, *c* is the speed of light in vacuum, *n* is the refractive index of the sample medium, and *h* is Planck's constant. Changes in shape between Fig. 4 (B and C) arise from the division by frequency when converting between optical density and absorption *B* coefficient spectrum in Eq. 14. For determination of the Stokes' shift, it is not necessary to know the quantum dot concentration to obtain the Einstein $\langle b \rangle$ lineshape. The lineshape can be normalized to its maximum peak height as in Fig. 4C. Alternatively, a lineshape normalized to unit area is needed to determine relative transition strengths from the 2D spectrum (see the Supplementary Materials).

If other absorption transitions contribute to the optical density spectrum, the absorption for the photoluminescent transition can be obtained from the optical density spectrum using an experimental procedure from (53). At a minimum, experimental measurement of the ensemble Stokes' shift (2λ) requires a maximum in this line-shape, although determination of $2(\Lambda - \lambda)$ in Eq. 10 will be distorted if the minimum on the high-frequency side is not separated from the maximum by the width of the static inhomogeneous lineshape. A zero baseline below the bandgap is required for use of Eq. 10.

PL spectra and conversion to Einstein stimulated emission B spectra

Steady-state PL spectra of covered and part-covered PbS nanocrystals were recorded on diluted nanocrystal solutions in tetrachloroethylene using an 850-nm excitation light source with 30-nm FWHM bandwidth (*54*). The part-covered quantum dot samples were diluted in tetrachloroethylene to $OD_{max}(1S - 1S) = 0.27$ at the bandgap in the 10-mm cuvette (the covered sample had $OD_{max}(1S - 1S) = 0.20$). PL quantum yields were 35% (ligand-covered) and 15% (part-covered). PL spectra were detected in a perpendicular or side-view geometry and recorded at a wavelength sampling interval of 5 nm; the spectrum from a nominally identical cuvette containing tetrachloroethylene was subtracted as a background.

After accounting for wavelength-dependent spectrograph sensitivity, the time-integrated emission spectrum, $F_{det}(\lambda)$, has units of photons per wavelength. The PL spectrum is then corrected for the inner-filter, self-absorption, or reabsorption effect (52). Assuming a

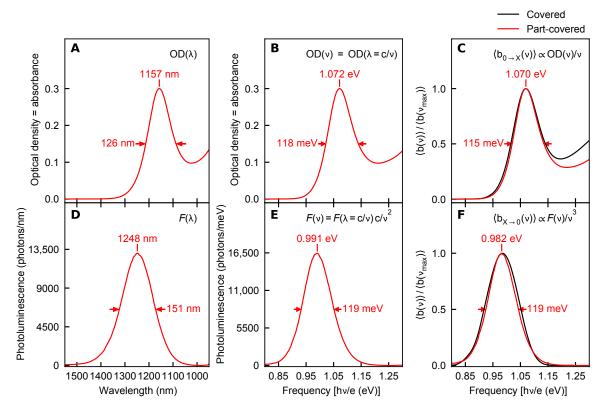


Fig. 4. Transformation of ensemble absorbance and PL spectra as a function of wavelength into absorption and stimulated emission Einstein *B* coefficient lineshapes as a function of frequency. The 1S-1S maximum and FWHM are marked in each panel. (A) Optical density (= decadic absorbance) spectrum as a function of wavelength. (B) Optical density spectrum as a function of frequency. (C) Einstein *B* absorption coefficient lineshape. (D) PL (= spontaneous emission) spectrum as a function of wavelength. (E) Spontaneous emission spectrum as a function of frequency $\langle a_{X \to 0}(v) \rangle \propto F(v)$. (F) Einstein *B* stimulated emission coefficient lineshape. The right column compares the ensemble-averaged Einstein *B* coefficient absorption lineshapes (C) and stimulated emission lineshapes (F) for the ligand-covered (black curve) and part-covered (red curve) PbS nanocrystals. Each lineshape is peak normalized.

narrow excitation beam in the center of the square cell, the perpendicularly detected emission spectrum is related to the true emission spectrum by equation 37 of (55)

$$F(\lambda) = F_{det}(\lambda) \, 10^{+OD(\lambda)/2} \tag{15}$$

Self-absorption red-shifts the detected PL maximum from its true position by \sim 4 to 5 meV. The corrected PL wavelength spectrum on the left-hand side of Eq. 15 is shown in Fig. 4D.

The time-integrated PL frequency spectrum is an irradiance spectrum and must be converted from wavelength to frequency using Eq. 12, so that

$$F(v) = -F(\lambda = c/v) d\lambda/dv = F(\lambda = c/v) c/v^2$$
(16)

The PL spectrum on the left-hand side of Eq. 16 has units of photons per frequency. Changes in shape between Fig. 4 (D and E) arise from both the nonlinear distortion of the horizontal axis (which also occurs between Fig. 4, A and B) and the frequency-dependent change of variables factor between wavelength and frequency in Eq. 16.

For homogeneous materials, rapid thermal quasi-equilibrium implies both that the emission spectrum is independent of the excitation wavelength (56) and that the Einstein A spontaneous emission spectrum is proportional to the time-integrated PL spectrum

$$a_{X \to 0}(\mathbf{v}, p, T) \propto F(\mathbf{v}) \tag{17}$$

The requirement of thermal quasi-equilibrium excludes partially equilibrated emission from excitons and defect states not at thermal quasi-equilibrium with each other, but quasi-equilibrium emission from either state alone will obey a generalized Einstein relation with its absorption. Heterogeneous materials require different considerations, addressed in the Supplementary Material, to guarantee that the ensemble average is undistorted in the time-integrated spectrum so that $\langle a_{j, X \to 0}(v, p, T) \rangle_{j} \propto F(v)$, which combined with Eq. 2A gives

$$\langle b_{j,X\to 0}(v,p,T) \rangle_j \propto F(v)/v^3$$
 (18)

Equation 2A shows that the shape of a spontaneous emission spectrum is distorted by a v^3 factor that does not appear in stimulated emission spectra. Changes in shape between Fig. 4 (E and F) arise from the resulting frequency-dependent conversion from spontaneous to stimulated emission lineshapes in Eq. 18. A zero baseline is needed on the high-frequency side of this lineshape if it is used in the generalized Einstein relation to predict the absorption lineshape.

The radiative lifetime

The spontaneous emission rate is given by Eq. 1C. Substituting the stimulated emission *b*-spectrum for the spontaneous emission *a*-spectrum using Eq. 2A gives

$$\Gamma_{j,X\to 0}^{a} = \int_{0}^{\infty} \frac{8\pi h v^{3} n^{3}}{c^{3}} b_{j,X\to 0}(v) \, dv \tag{19}$$

(The pressure and temperature arguments are suppressed in this subsection). Using the integrated Einstein *B* coefficient and area normalized lineshape $b_{j,X\to0}(v) = B_{j,X\to0}^v g_{j,X\to0}(v)$, Eq. 19 can be rewritten as

$$\Gamma_{j,X\to0}^{a} = \frac{8\pi h n^{3}}{c^{3}} B_{j,X\to0}^{\nu} \int_{0}^{\infty} \nu^{3} g_{j,X\to0}(\nu) \, d\nu \approx \frac{8\pi h n^{3} \nu_{X\to0}^{3}}{c^{3}} B_{j,X\to0}^{\nu}$$
(20)

Using Eq. 5, the integrated Einstein B coefficient can be rewritten as

$$B_{j,X\to0}^{\nu} = \left[\frac{B_{j,X\to0}^{\nu}}{B_{j,0\to X}^{\nu}}\right] B_{j,0\to X}^{\nu} \simeq \left[\frac{B_{j,X\to0}^{\nu}}{B_{j,0\to X}^{\nu}}\right] \int_{0}^{\infty} \frac{\sigma_{0\to X}(\nu) c}{h\nu n} d\nu$$
$$\approx \left[\frac{B_{X\to0}^{\nu}}{B_{0\to X}^{\nu}}\right] \frac{\sigma_{0\to X}^{\max} \Delta \nu_{0\to X} c}{h\nu_{0\to X} n}$$
(21)

where $\Delta v_{0 \rightarrow X}$ is the absorption linewidth and $\sigma_{0 \rightarrow X}^{max}$ is the absorption cross section maximum at $v_{0 \rightarrow X}$. The first approximation becomes an equality for homogeneously broadened spectra, in which case inserting Eq. 21 into Eq. 20 generalizes the Strickler-Berg relation (7) by eliminating the Condon approximation. Using both approximations gives a simple estimate of the radiative rate

$$\Gamma_{j,X\to0}^{a} \approx \left[\frac{w_{X\to0}}{w_{0\to X}}\right] \frac{8\pi n^2}{c^2} \frac{v_{X\to0}^3}{v_{0\to X}} \sigma_{0\to X}^{\max} \Delta v_{0\to X}$$
(22)

where the ratio of *B* coefficients in Eq. 21 has been replaced by the corresponding ratio of weights from Table 1. Using the absorption linewidth from Fig. 4C, $\sigma_{0\rightarrow X}^{max} = 4.18 \times 10^{-20} \text{ m}^2$ calculated from the molar extinction coefficient in (57), a solvent refractive index of n = 1.5, and the frequencies from Table 1, the simple estimate gives a radiative lifetime of $1/\Gamma_{j,X\rightarrow 0}^{a} \approx 2 \times 10^{-7}$ s for the bright exciton at internal thermal quasi-equilibrium. The ratio in brackets, which is not included in the Strickler-Berg relation (7), extends the radiative lifetime by about an order of magnitude. The actual lifetime can be shortened through quenching or lengthened through equilibrium with unquenched dark states.

2D spectra and conversion to Einstein B 2D spectra

A femtosecond Ti:sapphire regenerative amplifier operating at 1-kHz repetition rate pumps a noncollinear optical parametric amplifier (58) to generate the femtosecond short-wave infrared pulses used for 2D spectroscopy. The pulses are compressed down to ~16 fs by a deformable mirror grating compressor using an adaptive algorithm (58, 59). The pulses used for this experiment have energies of 3.6 µJ with a stability better than 0.4%. 2DFT spectra were recorded using the pump-probe geometry with the 2D spectrometer shown in fig. S2. The 2D spectrometer incorporates an actively stabilized Mach-Zehnder interferometer to generate the excitation pump-pulse pair (pulses *a* and *b*) with a delay, τ . The interferometer achieves a pathlength difference stability of less than ±0.6 nm. A noncollinear detection probe pulse *c* is used to measure the change in absorption after a waiting time T_w (60–62). The signal is modulated only in

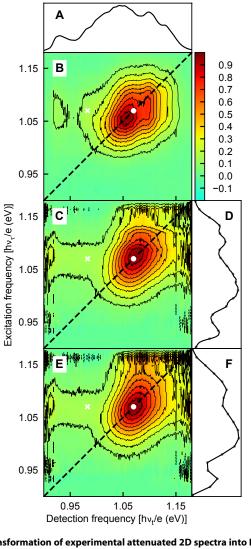


Fig. 5. Transformation of experimental attenuated 2D spectra into Einstein B 2D spectra. (A) Probe or detection pulse spectrum measured with the InGaAs spectrograph. (B) Real-valued 2D correlation spectrum of the part-covered sample at 1-ps waiting time. This attenuated 2D spectrum S_{2D} is filtered by the pulse spectra and distorted by sample absorption. (C) Einstein B 2D spectrum from Eq. 23 using division by the probe pulse spectrum along both frequency axes. (D) Transmitted probe pulse spectrum with frequency sampling points corresponding to sampling points of the 2D spectrum marked as circles. (**E**) Einstein *B* 2D spectrum S_{2D}^{β} from Eq. 23 using division by the spectrograph probe pulse spectrum along the detection axis and the interferogram-detected pump pulse-pair spectrum along the excitation axis. (F) Excitation pulse-pair spectrum from the Fourier transform of the excitation pulse-pair interferogram collected during the 2D experiment - frequency sampling points (circles) match the sampling points of the 2D spectrum. White dots on 2D spectra mark the 1S-1S absorption lineshape maximum; white diagonal crosses mark the 1S-1S stimulated emission lineshape maximum. Black contours are at 10% intervals and color changes are at 1% intervals (color bar at top right).

amplitude as a function of the time delay between the pump-pulse pair τ , which results in purely absorptive 2D spectra.

Nonlinear optical experiments on the sample of PbS nanocrystals were performed using a custom-built spinning sample cell (31). The sample temperature was 291 K under the ~84 kPa ambient atmospheric pressure in Boulder, CO. The optical density of the sample solutions was adjusted to around 0.3 in ~270- μ m pathlength. The

spinning sample cell keeps the sample air-free with ultra high vacuum compatible seals and completely exchanges the sample between laser shots. Using the beam diameter of 100 μ m, pulse energy of 10.2 nJ for each pump pulse in the pump pulse pair, pulse energy of 10.2 nJ for the probe pulse at the sample position, pulse spectrum, and the estimated frequency-dependent extinction coefficient (*57*), the average excitation probability was calculated to be ~9%, so over 90% of the signal arises from singly excited quantum dots.

The pump pulse-pair interferogram is recorded during the 2D scan and Fourier transformed to retrieve the pump pulse-pair spectrum $I_{ab}(v)$ (the absolute value of the complex-valued Fourier transform) and the spectral phase offset between pump pulses (its phase) (63). This also allows phase correction of the 2D spectra for both the Mach-Zehnder interferometer phase difference and the delay sampling offset without any adjustable parameters (see below). For each 2D spectrum, the waiting time T_w between the probe pulse c and the last pulse of the pump-pulse pair (a or b) is fixed. The probe pulse, c, and the signal are coupled into a single-mode fiber entering a spectrograph with an InGaAs detector array. The interference between the signal and probe pulse c is directly recorded as a function of the detection wavelength λ_t . The probe + signal interference spectrum is corrected for the wavelength sensitivity of the spectrograph to generate spectra with units of photons per pixel. The change of variables transformation in Eq. 12 generates probe + signal interference frequency spectra with units of photons per frequency. The excitation frequency (v_{τ}) axis is indirectly obtained by Fourier transforming the array of directly detected probe spectrum over a two-sided scan from positive to negative τ (60, 64). By using the 1D procedure for two-sided interferograms (63) at each detection frequency, the 2D spectrum is phase-corrected with the excitation frequency-dependent spectral phase offset between pump pulses obtained from the pump pulse-pair interferogram.

2D spectra of PbS quantum dots at waiting times from $T_w = 0$ to 500 fs evolve rapidly as a function of the waiting time. In particular, the stimulated emission Stokes' shift develops during this range of waiting times. Waiting times of 1 and 5 ps were chosen for this study because the shape of the 2D spectrum appeared stationary and because a 1-ps waiting time exceeds the reported exciton-spin relaxation times for similarly sized PbS quantum dots (*34, 35*), so the bright exciton was expected to be in thermal quasi-equilibrium.

In the partially collinear geometry shown in Fig. 1C, the experimentally measured 2DFT spectra with units of photons per (frequency)² are attenuated 2D spectra, denoted by $S_{2D}^-(v_b v_t; T_W)$ in (65). Attenuated 2D spectra are distorted by the photon number spectra of the excitation and detection pulses as they propagate through and are absorbed by the sample. After all coherence has decayed, the experimental 2D spectra can be exactly transformed to a dimensionless Einstein *B* representation that eliminates the propagation and absorptive distortions and the distortions from the pulse spectra

$$S_{2D}^{B}(v_{l}, v_{\tau}; T_{W}) = \frac{\ln(10) \text{ OD}(v_{\tau})}{I_{ab}(v_{\tau}) [1 - \exp(-\ln(10) \text{ OD}(v_{\tau}))]} \\ \cdot \frac{1}{I_{c}(v_{l}) \exp(-\ln(10) \text{ OD}(v_{l}))} \\ \cdot S_{2D}^{-}(v_{l}, v_{\tau}; T_{W})$$
(23)

where OD(v) is the linear optical density (decadic absorbance) spectrum of the sample from Eq. 13, and I_{ab} and I_c are pulse photon number spectra for the pump pulse-pair and the probe pulse, respectively

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(dimensions = photons per frequency) at the sample entrance. The term on the first line of the right-hand side of Eq. 23 transforms the 2D spectrum along the excitation frequency axis to remove the filtering effect of the pump pulse spectrum, including the attenuation of the pump pulse as it propagates through the sample. This attenuation factor is precisely division by the average of the pump pulse spectrum over the sample length (64-66). The term on the second line of the right-hand side of Eq. 23 transforms the 2D spectrum along the detection frequency axis to remove the combined filtering effect of the probe pulse spectrum, probe attenuation, and signal attenuation along the detection axis. Since attenuation of the probe has the same effect as attenuation of the signal, this attenuation factor is precisely division by the transmitted probe spectrum (64–66). [An equivalent last step for processing 2D spectra recorded with the fully noncollinear geometry would require division by the electric field spectra of separate pulses c (probe) and d (detection)—most 2D spectra are shown with only the latter division and thus depend on both the pulse spectra and the sample properties.] The resulting 2D spectrum on the left-hand side of Eq. 23 is called the Einstein B coefficient representation and goes beyond the ideal representation of 2D spectra (66) in dividing the ideal 2D spectrum by the pulse spectra at the sample entrance. Like the 1D absorption cross section or molar extinction coefficient, the Einstein B 2D spectrum is a microscopic property of the sample that is independent of the macroscopic pulse spectra and sample absorbance used to record it.

After this transformation, the 2D spectra can be modeled using only the Einstein b spectra in Eq. 7. An example 2D spectrum before and after conversion to the Einstein B representation is shown in Fig. 5. Division by the low-amplitude wings of the pulse spectra eventually diverges, so the data can only be fit within the region where the noise does not blow up appreciably. For the excitation frequency, the boundaries of this region are chosen as the half maxima of the pump pulse spectrum. For the detection frequency, the boundaries are chosen as the high-energy half maximum and the low-energy quarter maximum of the probe pulse spectrum. This unequal boundary (marked in Fig. 2) is due to the differences in resolution between the two axes and the contribution of emission signals along only the detection axis. The Einstein B 2D spectra are filtered outside this boundary region to minimize visual distraction in Figs. 2 and 5.

The pulse spectrum associated with each axis must be used when converting the attenuated representation of the 2D spectrum into the Einstein B 2D photon number spectrum. The excitation frequency axis is indirectly measured through the Fourier transform, whereas the detection frequency axis is directly resolved using the spectrograph. Unlike the excitation axis intensities, which only depend on accurate time delay control of the pump pulse pair, the detection axis intensities depend on coupling into the single-mode fiber in a consistent manner. Fine differences between pump and probe paths in the 2D interferometer make their pulse spectra slightly different. The spectra also differ slightly between the two pump arms of the Mach-Zehnder interferometer, as discussed in (67). Using probe pulse spectra to approximate pump spectra in Eq. 23 can generate artifacts in the Einstein B representation of the experimental 2D spectrum. As shown in Fig. 5C, this approximation causes an apparent kink across the 2D spectrum near $(hv_{\pi}/e) = 1.116$ eV excitation. Figure 5E shows the Einstein B 2D photon number spectrum calculated using the pump pulse-pair spectrum $I_{ab}(v_{\tau})$ from the Fourier transform of the pump pulse-pair interferogram measured during the 2D scan. The 2D spectrum is obviously smoother when the pump pulse-pair spectrum is used for division along the excitation frequency axis. This smoother 2D spectrum also leads to a quantitatively better model fit; the optimal reduced chi squared is $\chi_v^2 = 1.56$ when approximating the pump pulse-pair spectrum by the probe pulse spectrum versus 1.04 when using the pump pulse-pair spectrum.

SUPPLEMENTARY MATERIALS

Supplementary material for this article is available at http://advances.sciencemag.org/cgi/ content/full/7/22/eabf4741/DC1

REFERENCES AND NOTES

- 1. R. T. Ross, M. Calvin, Thermodynamics of light emission and free-energy storage in photosynthesis. *Biophys. J.* **7**, 595–614 (1967).
- R. E. Blankenship, *Molecular Mechanisms of Photosynthesis* (Blackwell Science Ltd., 2002).
 W. Shockley, H. J. Queisser, Detailed balance limit of efficiency of *p-n* junction solar cells.
- J. Appl. Phys. 32, 510–519 (1961).
 P. Würfel, U. Würfel, Physics of Solar Cells (Wiley-VCH, ed. 3, 2016).
- A. Einstein, in Sources of Quantum Mechanics, B. L. van der Waerden, Ed. (Dover, 1967), pp. 63–77.
- B. I. Stepanov, A universal relation between the absorption and luminescence spectra of complex molecules. Sov. Phys.-Dokl. 2, 81–84 (1957).
- S. J. Strickler, R. A. Berg, Relationship between absorption intensity and fluorescence lifetime of molecules. J. Chem. Phys. 37, 814–822 (1962).
- D. E. McCumber, Einstein relations connecting broadband emission and absorption spectra. *Phys. Rev.* 136, A954–A957 (1964).
- R. T. Ross, Radiative lifetime and thermodynamic potential of excited states. *Photochem. Photobiol.* 21, 401–406 (1975).
- R. L. Van Metter, R. S. Knox, On the relation between absorption and emission spectra of molecules in solution. *Chem. Phys.* 12, 333–340 (1976).
- 11. Y. B. Band, D. F. Heller, Relationships between the absorption and emission of light in multilevel systems. *Phys. Rev. A* **38**, 1885–1895 (1988).
- S. Park, T. Joo, Diffractive optics based three-pulse photon echo peak shift studies of spectral diffusion in polar liquids: Evidence for long lived frequency correlations. *J. Chem. Phys.* **131**, 164508 (2009).
- A. Hagfeldt, M. Graetzel, Light-induced redox reactions in nanocrystalline systems. *Chem. Rev.* 95, 49–68 (1995).
- B. H. Kim, J. H. Heo, S. Kim, C. F. Reboul, H. Chun, D. Kang, H. Bae, H. Hyun, J. Lim, H. Lee, B. Han, T. Hyeon, A. P. A. Alivisatos, P. Ercius, H. Elmlund, J. P. Park, Critical differences in 3D atomic structure of individual ligand-protected nanocrystals in solution. *Science* 368, 60–67 (2020).
- M. C. Weidman, M. E. Beck, R. S. Hoffman, F. Prins, W. A. Tisdale, Monodisperse, air-stable PbS nanocrystals via precursor stoichiometry control. ACS Nano 8, 6363–6371 (2014).
- M. P. Hendricks, M. P. Campos, G. Cleveland, I. Jen-La Plante, J. S. Owen, A tunable library of substituted thiourea precursors to metal sulfide nanocrystals. *Science* **348**, 1226–1230 (2015).
- J. R. Caram, S. N. Bertram, H. Utzat, W. R. Hess, J. A. Carr, T. S. Bischof, A. P. Beyler, M. W. B. Wilson, M. G. Bawendi, PbS nanocrystal emission is governed by multiple emissive states. *Nano Lett.* 16, 6070–6077 (2016).
- Y. Dong, T. Qiao, D. Kim, D. Parobek, D. Rossi, D. H. Son, Precise control of quantum confinement in cesium lead halide perovskite quantum dots via thermodynamic equilibrium. *Nano Lett.* 18, 3716–3722 (2018).
- S. D. Park, D. Baranov, J. Ryu, B. Cho, A. Halder, S. Seifert, S. Vajda, D. M. Jonas, Bandgap inhomogeneity of a PbSe quantum dot ensemble from two-dimensional spectroscopy and comparison to size inhomogeneity from electron microscopy. *Nano Lett.* **17**, 762–771 (2017).
- J. Cui, A. P. Beyler, I. Coropceanu, L. Cleary, T. R. Avila, Y. Chen, J. M. Cordero, S. L. Heathcote, D. K. Harris, O. Chen, J. Cao, M. G. Bawendi, Evolution of the single-nanocrystal photoluminescence linewidth with size and shell: Implications for exciton-phonon coupling and the optimization of spectral linewidths. *Nano Lett.* **16**, 289–296 (2016).
- J. D. Hybl, A. W. Albrecht, S. M. Gallagher Faeder, D. M. Jonas, Two-dimensional electronic spectroscopy. *Chem. Phys. Lett.* 297, 307–313 (1998).
- S. Mukamel, Multidimensional femtosecond correlation spectroscopies of electronic and vibrational excitations. Annu. Rev. Phys. Chem. 51, 691–729 (2000).
- D. M. Jonas, Two-dimensional femtosecond spectroscopy. Annu. Rev. Phys. Chem. 54, 425–463 (2003).
- 24. R. C. Hilborn, Einstein coefficients, cross sections, f values, dipole moments, and all that. *Am. J. Phys.* **50**, 982–986 (1982).

- V. Balevičius Jr., T. Wei, D. Di Tommaso, D. Abramavicius, J. Hauer, T. Polívka, C. D. P. Duffy, The full dynamics of energy relaxation in large organic molecules: From photo-excitation to solvent heating. *Chem. Sci.* **10**, 4792–4804 (2019).
- 26. J. G. Kirkwood, I. Oppenheim, Chemical Thermodynamics (McGraw-Hill, 1961).
- C. Cohen-Tannoudji, J. Dupont-Roc, G. Grynberg, Atom-Photon Interactions: Basic Processes and Applications (Wiley-Interscience, 1992).
- M. Planck, *The Theory of Heat Radiation* (Dover Press, ed. Translation of the Second (1912) German Edition, 1959).
- K. Bian, B. T. Richards, H. Yang, W. Bassett, F. W. Wise, Z. Wang, T. Hanrath, Optical properties of PbS nanocrystal quantum dots at ambient and elevated pressure. *Phys. Chem. Chem. Phys.* 16, 8515–8520 (2014).
- 30. C. Kittel, H. Kroemer, Thermal Physics (W. H. Freeman, ed. 2, 1980).
- D. Baranov, R. J. Hill, J. Ryu, S. D. Park, A. Huerta-Viga, A. R. Carollo, D. M. Jonas, Interferometrically stable, enclosed, spinning sample cell for spectroscopic experiments on air-sensitive samples. *Rev. Sci. Instrum.* 88, 014101 (2017).
- W. Qin, P. Guyot-Sionnest, Evidence for the role of holes in blinking: Negative and oxidized CdSe/CdS dots. ACS Nano 6, 9125–9132 (2012).
- 33. F. Fan, O. Voznyy, R. P. Sabatini, K. T. Bicanic, M. M. Adachi, J. R. McBride, K. R. Reid, Y.-S. Park, X. Li, A. Jain, R. Quintero-Bermudez, M. Saravanapavanantham, M. Liu, M. Korkusinski, P. Hawrylak, V. I. Klimov, S. J. Rosenthal, S. Hoogland, E. H. Sargent, Continuous-wave lasing in colloidal quantum dot solids enabled by facet-selective epitaxy. *Nature* **544**, 75–79 (2017).
- J. C. Johnson, K. A. Gerth, Q. Song, J. E. Murphy, A. J. Nozik, G. D. Scholes, Ultrafast exciton fine structure relaxation dynamics in lead chalcogenide nanocrystals. *Nano Lett.* 8, 1374–1381 (2008).
- F. Masia, W. Langbein, I. Moreels, Z. Hens, P. Borri, Exciton dephasing in lead sulfide quantum dots by X-point phonons. *Phys. Rev. B* 83, 201309 (2011).
- F. Gesuele, M. Y. Sfeir, W. K. Koh, C. B. Murray, T. F. Heinz, C. W. Wong, Ultrafast supercontinuum spectroscopy of carrier multiplication and biexcitonic effects in excited states of PbS quantum dots. *Nano Lett.* **12**, 2658–2664 (2012).
- J. M. An, A. Franceschetti, A. Zunger, The excitonic exchange splitting and radiative lifetime in PbSe quantum dots. *Nano Lett.* 7, 2129–2135 (2007).
- Z. Hu, Y. Kim, S. Krishnamurthy, I. D. Avdee, M. O. Nestoklon, A. S. Singh, A. V. Malko, S. V. Goupalov, J. A. Hollingsworth, H. Htoon, Intrinsic exciton photophysics of PbS quantum dots revealed by low-temperature single nanocrystal spectroscopy. *Nano Lett.* 19, 8519–8525 (2019).
- M. T. Trinh, M. Y. Sfeir, J. J. Choi, J. S. Owen, X. Zhu, A hot electron-hole pair breaks the symmetry of a semiconductor quantum dot. *Nano Lett.* 13, 6091–6097 (2013).
- T. J. Liptay, L. F. Marshall, P. S. Rao, R. J. Ram, M. G. Bawendi, Anomalous Stokes shift in CdSe nanocrystals. *Phys. Rev. B* 76, 155314 (2007).
- A. N. Poddubny, M. O. Nestoklon, S. V. Goupalov, Anomalous suppression of valley splittings in lead salt nanocrystals without inversion center. *Phys. Rev. B* 86, 035324 (2012).
- W. Langbein, J. M. Hvam, Biexcitonic bound and continuum states of homogeneously and inhomogeneously broadened exciton resonances. *Phys. Status Solidi A* **190**, 167–174 (2002).
- T. Ghosh, J. Dehnel, M. Fabian, E. Lifshitz, R. Baer, S. Ruhman, Spin blockades to relaxation of hot multiexcitons in nanocrystals. J. Phys. Chem. Lett. 10, 2341–2348 (2019).
- M. S. Azzaro, A. K. Le, H. Wang, S. T. Roberts, Ligand-enhanced energy transport in nanocrystal solids viewed with two-dimensional electronic spectroscopy. J. Phys. Chem. Lett. 10, 5602–5608 (2019).
- T. Förster, Fluoreszenzspektrum und Wasserstoffionen-konzentration. Naturwissenschaften 36, 186–187 (1949).
- R. A. Marcus, Relation between charge transfer absorption and fluorescence spectra and the inverted region. J. Phys. Chem. 93, 3078–3086 (1989).
- 47. S. R. de Groot, P. Mazur, Non-Equilibrium Thermodynamics (Dover, 1984).
- G. D. Scholes, G. R. Fleming, L. X. Chen, A. Aspuru-Guzik, A. Buchleitner, D. F. Coker, G. S. Engel, R. van Grondelle, A. Ishizaki, D. M. Jonas, J. S. Lundeen, J. K. McCusker, S. Mukamel, J. P. Ogilvie, A. Olaya-Castro, M. A. Ratner, F. C. Spano, K. B. Whaley, X. Zhu, Utilizing coherence to enhance function in chemical and biophysical systems. *Nature* 543, 647–656 (2017).
- D. M. Jonas, Vibrational and nonadiabatic coherence in 2D electronic spectroscopy, the Jahn-Teller effect, and energy transfer. Annu. Rev. Phys. Chem. 69, 327–352 (2018).
- J. Cao, R. J. Cogdell, D. F. Coker, H.-G. Duan, J. Hauer, U. Kleinekathöfer, T. L. C. Jansen, T. Mančal, R. J. D. Miller, J. P. Ogilvie, V. I. Prokhorenko, T. Renger, H.-S. Tan, R. Tempelaar, M. Thorwart, E. Thyrhaug, S. Westenhoff, D. Zigmantas, Quantum biology revisited. *Sci. Adv.* 6, eaaz4888 (2020).
- D. Baranov, M. J. Lynch, A. C. Curtis, A. R. Carollo, C. R. Douglass, A. M. Mateo-Tejada, D. M. Jonas, Purification of oleylamine for materials synthesis and spectroscopic diagnostics for *trans* isomers. *Chem. Mater.* **31**, 1223–1230 (2019).
- 52. J. D. Ingle, S. R. Crouch, Spectrochemical Analysis (Prentice-Hall, 1988).

- I. Ketskeméty, J. Dombi, R. Horvai, Fluoreszenzemission, Absorption und Temperaturstrahlung von Lösungen. Ann. Phys. 463, 342–352 (1961).
- O. E. Semonin, J. C. Johnson, J. M. Luther, A. G. Midgett, A. J. Nozik, M. C. Beard, Absolute photoluminescence quantum yields of IR-26 Dye, PbS, and PbSe quantum dots. *J. Phys. Chem. Lett.* 1, 2445–2450 (2010).
- G. A. Crosby, J. N. Demas, Measurement of photoluminescence quantum yields. Review. J. Phys. Chem. 75, 991–1024 (1971).
- E. H. Kennard, On the interaction of radiation with matter and on fluorescent exciting power. *Phys. Rev.* 28, 672–683 (1926).
- I. Moreels, K. Lambert, D. Smeets, D. De Muynck, T. Nollet, J. C. Martins, F. Vanhaecke, A. Vantomme, C. Delerue, G. Allan, Z. Hens, Size-dependent optical properties of colloidal PbS quantum dots. ACS Nano 3, 3023–3030 (2009).
- D. Brida, S. Bonora, C. Manzoni, M. Marangoni, P. Villoresi, S. De Silvestri, G. Cerullo, Generation of 8.5-fs pulses at 1.3 μm for ultrabroadband pump-probe spectroscopy. *Opt. Express* 17, 12510–12515 (2009).
- E. Zeek, K. Maginnis, S. Backus, U. Russek, M. Murnane, G. Mourou, H. Kapteyn, G. Vdovin, Pulse compression by use of deformable mirrors. *Opt. Lett.* 24, 493–495 (1999).
- S. M. Gallagher Faeder, D. M. Jonas, Two-dimensional electronic correlation and relaxation spectra: Theory and model calculations. J. Phys. Chem. A 103, 10489–10505 (1999).
- E. M. Grumstrup, S.-H. Shim, M. A. Montgomery, N. H. Damrauer, M. T. Zanni, Facile collection of two-dimensional electronic spectra using femtosecond pulse-shaping technology. *Opt. Express* 15, 16681–16689 (2007).
- F. D. Fuller, J. P. Ogilvie, Experimental implementations of two-dimensional Fourier transform electronic spectroscopy. Annu. Rev. Phys. Chem. 66, 667–690 (2015).
- 63. R. J. Bell, Introductory Fourier Transform Spectroscopy (Academic Press, 1972).
- J. D. Hybl, A. Albrecht Ferro, D. M. Jonas, Two-dimensional Fourier transform electronic spectroscopy. J. Chem. Phys. 115, 6606–6622 (2001).
- M. K. Yetzbacher, N. Belabas, K. A. Kitney, D. M. Jonas, Propagation, beam geometry, and detection distortions of peak shapes in two-dimensional Fourier transform spectra. *J. Chem. Phys.* **126**, 044511 (2007).
- A. P. Spencer, H. Li, S. T. Cundiff, D. M. Jonas, Pulse propagation effects in optical 2D Fourier-transform spectroscopy: Theory. J. Phys. Chem. A 119, 3936–3960 (2015).
- 67. J. Ryu, thesis, University of Colorado, Boulder (2018).
- N. C. Anderson, M. P. Hendricks, J. J. Choi, J. S. Owen, Ligand exchange and the stoichiometry of metal chalcogenide nanocrystals: Spectroscopic observation offacile metal-carboxylate displacement and binding. J. Am. Chem. Soc. 135, 18536–18548 (2013).
- D. Segets, J. M. Lucas, R. N. Klupp Taylor, M. Scheele, H. Zheng, A. P. Alivisatos, W. Peukert, Determination of the quantum dot band gap dependence on particle size from optical absorbance and transmission electron microscopy measurements. ACS Nano 6, 9021–9032 (2012).
- R. Ihly, J. Tolentino, Y. Liu, M. Gibbs, M. Law, The photothermal stability of PbS quantum dot solids. ACS Nano 5, 8175–8186 (2011).
- 71. M. Chemnitz, thesis, Friedrich-Schiller Universität Jena (2019).
- O. I. Micic, H. M. Cheong, H. Fu, A. Zunger, J. R. Sprague, A. Mascarenhas, A. J. Nozik, Size-dependent spectroscopy of InP quantum dots. J. Phys. Chem. B 101, 4904–4912 (1997).
- T. Förster, in *Biological Physics*, E. V. Mielczarek, E. Greenbaum, R. S. Knox, Eds. (American Institute of Physics, 1993), pp. 148–160.
- T. Förster, in *Modern Quantum Chemistry*, O. Sinanoğlu, Ed. (Academic Press Inc., 1965), vol. III, pp. 93–137.

- S. E. Braslavsky, E. Fron, H. B. Rodríguez, E. San Román, G. D. Scholes, G. Schweitzer, B. Valeur, J. Wirz, Pitfalls and limitations in the practical use of Förster's theory of resonance energy transfer. *Photochem. Photobiol. Sci.* 7, 1444–1448 (2008).
- R. J. Hill, T. L. Courtney, S. D. Park, D. M. Jonas, Lightweight hollow rooftop mirrors for stabilized interferometry. *Opt. Eng.* 52, 105103–105103 (2013).
- D. J. Jones, E. O. Potma, J.-x. Cheng, B. Burfeindt, Y. Pang, J. Ye, X. S. Xie, Synchronization of two passively mode-locked, picosecond lasers within 20 fs for coherent anti-Stokes Raman scattering microscopy. *Rev. Sci. Instrum.* **73**, 2843–2848 (2002).
- R. N. Bracewell, The Fourier Transform and its Applications (McGraw-Hill series in Electrical and Computer Engineering Circuits and Systems, McGraw Hill, ed. 3, 2000).
- S. T. Roberts, J. J. Loparo, A. Tokmakoff, Characterization of spectral diffusion from two-dimensional line shapes. J. Chem. Phys. 125, 084502 (2006).
- 80. P. A. Jansson, Deconvolution: With Applications in Spectroscopy (Academic Press, 1984).
- Y. Choi, S. Sim, S. C. Lim, Y. H. Lee, H. Choi, Ultrafast biexciton spectroscopy in semiconductor quantum dots: Evidence for early emergence of multiple-exciton generation. *Sci. Rep.* **3**, 3206 (2013).
- P. R. Bevington, Data Reduction and Error Analysis for the Physical Sciences (McGraw-Hill, ed. 3, 2003).
- 83. A. Rohatgi, "WebPlotDigitizer" version 4.4 (2020); https://automeris.io/WebPlotDigitizer.
- S. Mukamel, Femtosecond optical spectroscopy: A direct look at elementary chemical events. Annu. Rev. Phys. Chem. 41, 647–681 (1990).

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Relations between absorption, emission, and excited state chemical potentials from nanocrystal 2D spectra

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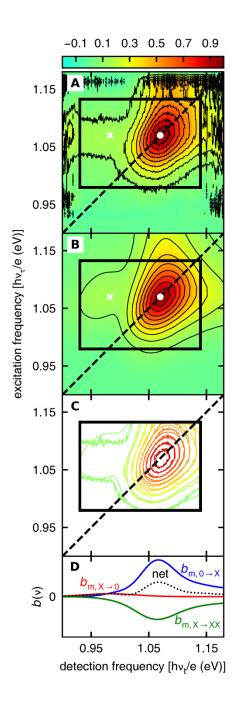
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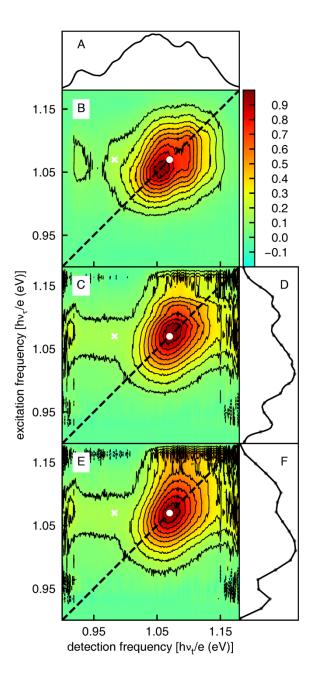
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high resolution Figure 5

high-resolution Figure 2



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Supplementary Materials for

Relations between absorption, emission, and excited state chemical potentials from nanocrystal 2D spectra

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The PDF file includes:

Materials Methods Supplementary Text Figs. S1 to S8 Tables S1 to S3 Legend for data set S1 References

Other Supplementary Material for this manuscript includes the following:

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Data set S1

Materials

Ligand-Covered Nanocrystal Synthesis

The sample of lead sulfide (PbS) nanocrystals with lowest electronic absorption peak at 1.074 eV (1154 nm) was synthesized from lead (II) chloride and elemental sulfur in oleylamine following the hot injection procedure reported by the Tisdale group(15) with modifications in the washing procedure, as described previously in detail.(51) Briefly, the synthesis was carried out by injecting 2 mL of sulfur solution in oleylamine (6 mg/ml concentration) into a saturated solution/suspension of lead (II) chloride (0.840 g) and oleylamine (7.5 mL) at 90.5 °C. Three minutes after the injection, the dark-brown reaction mixture was quenched by swiftly injecting room temperature anhydrous toluene followed by transferring the flask into an ice-filled water bath. Further post-synthetic handling, which included addition of oleic acid to the nanocrystals and their washing with anhydrous acetonitrile, were performed inside the argon-filled glovebox. ¹H NMR in deuterated benzene (C_6D_6) with an added internal standard ferrocene was used to determine total concentration of the ligands (free/bound oleic acid, oleate anion, and residual oleylamine) present in the nanocrystal sample.(68) The sizing curve and a molar extinction coefficient of PbS from ref. (57) were used to estimate the nanocrystal volume and concentration. Assuming a spherical nanocrystal shape, the nominal ligand surface coverage was found to be $\sim 3.1/nm^2$.

Part-covered nanocrystal synthesis

The sample of lead sulfide (PbS) nanocrystals with the lowest electronic absorbance peak at 1.072 eV (1157 nm) was synthesized according to a previously published procedure.(16) Briefly, a solution of N-(3,5-bistrifluoromethyl)Phenyl-N'-dodecyl thiourea (4.17 mmol) and diglyme (5ml) is injected into a solution of lead oleate (6.27 mmol) dissolved in 1-octene (134 mL) at 120 °C. The reaction is stirred for 30 seconds, cooled and purified by several rounds of precipitation/centrifugation/redispersion using toluene and methyl acetate. Partial removal of the oleate capping ligand was accomplished with 1,2-bis(dimethylphosphino)ethane. 1,2bis(dimethylphosphino)ethane (1 mL) is added to a toluene solution of lead sulfide nanocrystals (2.5 mL, 220 µM) and stirred for 10 minutes. After this time, the nanocrystals were precipitated by adding methyl acetate (15 mL) and the nanocrystals isolated by centrifugation. The pellet is then dissolved in tetrachloroethylene (2mL) and precipitated with methyl acetate (6ml) and the precipitate separated by centrifugation. The resulting nanocrystals were dissolved in pentane, transferred to a vial for storage, thoroughly dried under vacuum, and redissolved in anhydrous toluene to make a stock solution. The concentration of nanocrystals and ligands in this stock solution was determined by removing a known volume using a microliter syringe, drying it under vacuum before quantitative dilution in tetrachloroethylene for UV-Vis-NIR analysis or benzene d_6 for NMR analysis in a similar way as for the covered sample. Assuming a spherical nanocrystal shape, the nominal oleate surface coverage was found to be $\sim 1.09/nm^2$.

Electron microscopy analysis

Figure S1 shows Annular Dark Field – Scanning Transmission Electron Microscopy (ADF-STEM) images of "covered" and "part-covered" samples side by side. The results of ImageJ thresholding analysis(*19, 69*) are correlated for individual particles using histograms in Fig. S1 and summarized in Table S1. The average area equivalent circular diameter is very similar between two samples – 4.27(9) and 4.29(11) nm for part-covered and covered samples, respectively – but the standard deviation is ~30% larger for the covered sample. In the case of

the ellipse fit to the nanoparticle projections, the part-covered sample's major/minor axis projection ratio is larger (\sim 1.26) than that of the covered sample (\sim 1.14) with similar standard deviations of major and minor projections for both samples. Imaged nanocrystals in both samples are faceted and non-spherical. The conventional treatment of their projections as circles is not applicable and overestimates the standard deviation for the covered sample but is included here for comparison to literature results. Elliptical projection analysis in Fig. S1 shows the standard deviation for each dimension is similar across the two samples.

1D linear absorption and emission lineshapes

Linear absorption spectra of covered and part-covered PbS nanocrystals were recorded over the 400-1500 nm wavelength range with a 1 nm wavelength sampling interval using a Cary 5000 UV/Vis/NIR spectrophotometer. Absorbance(52) was measured in both the spinning sample cell used for 2D spectroscopy (0.27 mm pathlength) and the cuvette (10 mm pathlength) used for photoluminescence (PL) spectroscopy. For ligand-covered quantum dots, a small portion of dry nanocrystals was dissolved in tetrachloroethylene (Aldrich, anhydrous, $\geq 99\%$) and transferred into an air-tight cell for these measurements. A nominally identical sample cell (spinning or cuvette) was filled with tetrachloroethylene as a reference (absorption blank or PL background). The part-covered quantum dot solution was diluted in tetrachloroethylene for PL measurements. For deconvolution and application of the generalized Einstein relations, scans were repeated and averaged in order to improve the signal to noise ratio. Absorption and PL spectra can be distorted by sample aging from oxidation or photothermal instability.(70) To minimize this effect, the PL, absorption, and 2D Fourier Transform (2DFT) experiments should be performed consecutively. For the part-covered sample, the 2DFT and absorption measurements were taken on the same day, and the PL spectrum was measured 8 days later. After 8 days, the absorbance peak had red-shifted by 4 meV and its ensemble FWHM had narrowed by 2 meV; these changes fit into a longer term trend.

Steady-state PL spectra of covered and part-covered PbS nanocrystals were recorded on diluted nanocrystal solutions in tetrachloroethylene at the National Renewable Energy Laboratory. The customized Horiba spectrofluorometer used an 850 nm excitation light source with 30 nm FWHM bandwidth (Thorlabs M850L3 LED).(54) For these measurements, 10 mm square cuvettes with 4 polished faces made of Suprasil 300 fused silica (Starna cells) were modified to add an air-tight seal. The part-covered quantum dot samples were diluted in tetrchloroethylene to $OD_{max}(1S-1S) = 0.27$ at the bandgap in the 10 mm cuvette (the covered sample had $OD_{max}(1S-1S) = 0.20$). PL quantum yields were 35% (ligand covered) and 15% (part-covered). PL spectra were detected in a side-view geometry and recorded at a wavelength sampling interval of 5 nm; the spectrum from a nominally identical cuvette containing tetrachloroethylene solvent(71) is constant to within 1% over the 1S-1S absorption/emission wavelength range, so refractive index correction of the PL spectra(55) is not needed. After correction for wavelength dependent spectrograph sensitivity, the time-integrated emission spectrum, $F_{det}(\lambda)$, has units of photons/wavelength.

Eq. (16) assumes a PL spectrum with dimensions of photons/wavelength; dimensions of energy/wavelength (or power/wavelength), require additional division by the energy per photon hv, yielding an overall v^{-6} [instead of v^{-5} from Eq. (16)-(18)]. A sensitivity calibration curve yielding a spectrum *F* proportional to detected photocurrent (or the photovoltage that the

photocurrent generates across an impedance) would require overall multiplication by v^{-5} as in Eq. (16). Alternatively, a sensitivity calibration curve yielding a spectrum proportional to incident power (or incident energy over a time interval) would require an overall multiplication by v^{-6} to generate an Einstein *B* coefficient lineshape. As a point of reference, the different factors have lineshape maxima shifted by 2 meV for the sample studied here. The number of photons per unit wavelength in the lower left panel of Fig. 4 was estimated from the shot noise on the photoluminescence spectrum in order to illustrate the change of variables transformation to photons per unit frequency in the lower middle panel. Any multiplicative error in this estimate would not affect the normalized stimulated emission lineshape.

Three criteria must be satisfied for proportionality between the ensemble-averaged spontaneous emission rate spectrum and the time-integrated photoluminescence spectrum underlying Eq. (18): size-selective excitation of quantum dots(72) must be avoided by appropriate choice of excitation wavelength; excitation energy transfer(52) between dots must be avoided by measuring PL at low quantum dot concentration; and the quantum dot sample must not have a systematic correlation between photoluminescence quantum yield and bandgap. If the quantum yield varies appreciably, this third possibility can impact Eq. (18) without affecting the generalized Einstein relations in Eq. (2). The effect of such quantum yield inhomogeneity may or may not be compatible with misinterpretation as size-dispersion inhomogeneity. It can be detected by measuring the stimulated emission spectrum or the rate of spontaneous emission as a function of the frequency of the emitted light [for example, by time-resolved fluorescence upconversion].

For a heterogeneous ensemble, Förster resonance energy transfer (FRET) contributes another possible distortion of the PL spectrum.(52) FRET is a non-radiative transfer of excitation energy that proceeds through near-field virtual donor emission and acceptor absorption.(73, 74) Long range FRET is most probable when the donor emission spectrum overlaps the acceptor absorption spectrum. Significant FRET would red-shift the ensemble PL spectra in a PbS QD sample with bandgap inhomogeneities. In the limit of fast FRET compared to the exciton lifetime, excitations would equilibrate across quantum dots in the ensemble, so that ensemble absorption and emission would obey the homogeneous generalized Einstein relations.

The calculation below estimates an upper bound on the fraction of excited quantum dots that undergo FRET before spontaneous emission. This conventional estimate of the FRET rate does not include reduction of the exciton emission rate from degeneracy or the exciton chemical potential. For the part-covered sample with $OD_{max} = 0.27$ at the 1S-1S peak, pathlength of 10 mm, average nanocrystal diameter of 4.3 nm, and the literature molar decadic extinction coefficient maximum of $\varepsilon_{max}(1S-1S) = 1.09 \times 10^5/(M \cdot cm)$ from ref. (57) ($\sigma_{0\to X}^{max} = 4.18 \times 10^{-16}$ cm²), the nanocrystal concentration is $C = 2.48 \times 10^{-6}$ M ($N_0 = 1.49 \times 10^{15}/cm^3$). Using Eq. (5) of ref. (75), the perfect spectral overlap integral for Gaussian spectra is $J = (1/\sqrt{2})\varepsilon_{max}(1S-1S)/(\tilde{v}^4) \approx 1.56 \times 10^{-8}$ cm⁶/mol (for $\tilde{v} = 8389$ cm⁻¹). Taking the orientational factor $\kappa^2 = 1$ (isotropic absorbers), the emission quantum yield $\Phi_D = 1$, and the quantum dot solution refractive index n = 1.5, Eq. (4) of ref. (75) gives the Förster critical radius as $R_0^6 \approx$ 2.70×10^{-36} cm⁶. In dilute solutions, one must account for the much higher rate of FRET to the randomly located nearest neighbor, and the Förster critical concentration is a more important parameter.(73) The Förster critical concentration given by Eq. (20) of ref. (73) is $C_0 \approx$ 2.41×10^{-4} M. At concentrations well below the critical concentration, the PL quantum yield is equal to the PL quantum yield at infinite dilution, and Förster's dimensionless concentration parameter [Eq. (22) of ref. (73)] becomes $\gamma = C/C_0 \approx 0.010$. For $\gamma \ll 1$, Eq. (23) of ref. (73), which gives the fraction of PL from the primary absorber, can be simplified to $1 - \gamma \cdot (\pi/2\sqrt{2}) \approx$ 0.99. Thus, with a quantum dot concentration less than 1% of the Förster critical concentration, FRET contributes only a small distortion of less than 1% to the ensemble PL spectrum.

Other distortions of emission spectra are known to arise from the static size distribution of quantum dots. These effects are greatly reduced when the static broadening is comparable to or smaller than the dynamical linewidth. Size-selective excitation is avoided by exciting above the 1S-1S exciton transition.(72) For PbS quantum dots, the absorption cross section is approximately proportional to nanocrystal volume; (57) this generates size-bias in the excitation probability that can distort the ensemble PL spectrum.(40) The size distributions studied here have a $\pm 10\%$ size-bias in the excitation probability, which generates negligible changes in width and decreases the Stokes' shift by 1 meV. Based on Eq. (8), the $\pm 10\%$ variation in Stokes' shift with bandgap (deduced from Fig. S8) leads to a 5% variation in dynamical linewidth with bandgap, which also has a small effect on the ensemble emission spectrum. However, under a Gaussian approximation for absorption, the variation in Stokes' shift with bandgap is calculated to narrow the ensemble emission spectrum by $\sim 6\%$ without shifting it. Qualitatively, this narrowing is readily understood because larger bandgap dots have larger Stokes' shifts, which move their emission towards line center, yet smaller bandgap dots have smaller Stokes' shifts, which also move their emission towards line center. Although size-dependent intra-band splittings would be needed for a quantitative calculation in PbS quantum dots, the Gaussian approximation suggests this may account for the slight ensemble linewidth discrepancy between fit and experiment in Fig. 3B.

Methods

Sample preparation for 2D experiments

Nonlinear optical experiments on the sample of PbS nanocrystals were performed using a spinning sample cell that was custom built in our laboratory.(*31*) The spinning sample cell consists of an air-tight copper gasket sealed enclosure carrying a 4200 rpm, 2.5" hard drive disk motor and a reusable glass sandwich cell. Inside the glovebox, the glass sandwich cell was filled with the solution of PbS nanocrystals in tetrachloroethylene using a mechanical pipette and sealed in the airtight enclosure. The optical density of the sample solution was adjusted to around 0.3 [OD_{max}(1S-1S) =0.35 at 1154 nm for the part-covered sample and OD_{max}(1S-1S) = 0.31 for the ligand-covered sample]. The pathlength of the glass sandwich cell was measured to be ~270 µm with <5% min-max internal pathlength variation. The spinning sample cell keeps the sample air-free with ultra high vacuum compatible seals and completely exchanges the sample between laser shots. The re-sampling time is 0.9 s under illumination at 1 kHz because the laser and spin frequencies, measured with a common sampling clock, have no prime factor in common at the 0.01 Hz level. Cumulative centrifugation is prevented by stopping spinning every 20 minutes, allowing mixing by an inert gas bubble.(*31*)

2D spectroscopy experiments

2DFT spectra were recorded using the pump-probe geometry. In this partially collinear geometry, pump pulses a and b travel collinearly to create a spatially uniform frequency dependent excitation and the probe pulse c detects the change in absorption after the waiting

time.(60-62) The signal is modulated only in amplitude as a function of the time delay between the pump-pulse pair τ , which results in purely absorptive 2D spectra. Since a Fourier transform is a linear operation, the total 2D spectrum of an inhomogeneously broadened sample is a summation of the individual homogeneous 2D spectra.

A femtosecond Ti:sapphire regenerative amplifier operating at 1 kHz repetition rate (Spectra-Physics Spitfire Pro) pumps a non-collinear optical parametric amplifier (NOPA) to generate the femtosecond short-wave infrared (SWIR) pulses used for 2D spectroscopy. The NOPA is based on the design of Cerullo and co-workers,(58) and uses quasi-phase matching in a periodically poled stoichiometric lithium tantalate crystal to generate pulses over the $1 - 2 \mu m$ range. The pulses used for this experiment have energies of 3.6 μ J with a stability better than 0.4% directly out of the NOPA. The pulse spectra in the bottom panel of Fig. 2 have half maxima at 0:98 eV and 1:14 eV. The pulses are compressed down to ~16 fs at the sample position in the 2D interferometer by using a deformable mirror grating compressor and an adaptive algorithm(58, 59) to maximize the intensity of the second harmonic generation. The second harmonic is generated in a 20 μ m thick Type I BBO crystal cut at 20 degrees with respect to its optical axis. To match dispersion, a window identical to the front window of the spinning sample cell is inserted before the BBO crystal.

The 2D spectrometer is shown in Figure S2. The 2D spectrometer consists of an actively stabilized Mach-Zehnder interferometer for generating the pump-pulse pair (pulses a and b) with a delay, τ . A piezoelectric transducer attached to the lightweight dihedral retroreflector(76) in arm b of the interferometer finely adjusts and stabilizes the delay with feedback (76, 77) from a linearly polarized, continuous wave, red HeNe laser. A linearly polarized, continuous wave, yellow HeNe laser is also used to provide an out-of-loop check on stabilization. The interferometer achieves a pathlength difference stability of at least ± 0.6 nm. The loop filter was modified to change the polarity of the locking position to lock at both positive and negative slope zero-crossings of the red HeNe interferometer output. The positive and negative slope lock time series are each evenly spaced, but interleaving them would generate a slightly unequal spacing between alternate data points. This problem is solved by using the interlaced Fourier transform, which requires precise knowledge of the time offset between the positive and negative slope lock time series. (67, 78) The offset is determined to be 1.054 ± 0.001 fs by fitting the two aliased yellow HeNe interferograms. As a result, the two individual aliased spectra are combined into a single spectrum with twice the bandwidth. The pump-pulse pair and probe pulse are vertically displaced before focusing onto the sample. After transmission through the sample, the pumppulse pair is picked off and directed onto a photodiode to record the pump pulse-pair interferogram during the 2D scan.

RMS noise estimates for the 2D spectra

Two methods were used to estimate noise and errors in the 2D spectrum. Excitation pulse fluctuations and the detector array generate random noise. This random noise is estimated from the residual imaginary component of a 2D spectrum. Since 2D spectra measured with the pump-probe geometry should be purely real in the absence of noise, the Fourier amplitude in the imaginary 2D spectrum can be attributed to noise. The root mean square imaginary component within the rectangular fit region was, as a fraction of the maximum real 2D amplitude, 1.9% for the part-covered sample at $T_w = 1$ ps, 2.4% for the part-covered sample at $T_w = 5$ ps, and 4.1% for the ligand-covered sample at $T_w = 1$ ps. The error estimate is derived from the difference between two 2D spectra at long T_w , which should have the same shape. Each 2D experiment was set up independently. This requires 3 resets: pump-probe time zero, pulse spatial overlap, and probe

beam coupling into the single-mode fiber to the spectrograph. This difference includes potential sources of systematic error. Comparing the difference between the real part of the 2D spectra for the part-covered sample at $T_w = 1$ ps and 5 ps over the rectangular fit region gives a 2.4% root mean square error.

Equations for fitting 2D spectra

The strategy for simulating 2D spectra at long waiting times relies on the fact that the 2D spectrum for an ensemble is the sum of 2D spectra for the members of the ensemble. This diagonally elongates the 2D spectrum. For an isolated transition, the ellipticity of the 2D peakshape is widely used to separate static and dynamical broadening - static broadening is measured after spectral diffusion is complete.(79) The ellipticity is useful but not quantitative for overlapping 2D peaks. Here, an algorithm was developed to least-squares fit the 2D spectra to a model containing the three signal contributions in Fig. 1B. The algorithm (flowchart in Fig. S3) finds dynamical lineshapes in Eq. (7) by deconvolving the ensemble absorption and emission lineshapes with a trial Gaussian static bandgap distribution with standard deviation Δ_{static} .(19) Deconvolution neglects changes in the Stokes' shift, dynamical linewidth, and absorption cross section with bandgap, which are small if the bandgap distribution is narrow. This deconvolution starts with ensemble absorption and emission lineshapes.

The ensemble averaged Einstein B coefficient is

$$\left\langle B_{j,0\to X}^{\nu} \right\rangle_{j} = \int_{0}^{\infty} \left\langle b_{j,0\to X}(\nu) \right\rangle_{j} d\nu$$

$$\approx 2 \int_{\text{below}}^{\text{max}} \left\langle b_{j,0\to X}(\nu) \right\rangle_{j} d\nu$$

$$S(1)$$

where "below" is some frequency below the onset of $0 \rightarrow X$ absorption and "max" = $v_{0 \rightarrow X}^{ens}$ is the frequency of the first maximum in the ensemble-averaged absorption- $\langle b \rangle$. The approximation on the second line can be useful if the high frequency side of the absorption is overlapped by other transitions (see below). The ensemble absorption lineshape is given by

$$g_{0\to X}^{ens}(\nu) = \left\langle b_{j,0\to X}(\nu) \right\rangle_j / \left\langle B_{j,0\to X}^{\nu} \right\rangle_j \quad .$$
 S(2)a

The ensemble stimulated emission lineshape is defined by analogy to the ensemble absorption lineshape

$$g_{X\to 0}^{ens}(\nu) = \left\langle b_{j,X\to 0}(\nu) \right\rangle_j / \left\langle B_{j,X\to 0}^{\nu} \right\rangle_j.$$
 S(2)b

Proportionality constants in Eq. (17) and (18) cancel, and there is usually no difficulty integrating over the entire ensemble emission spectrum. A zero baseline is needed on both sides of this lineshape if it is used in the generalized Einstein relation to predict the absorption lineshape.

For each of the three processes that contribute to the 2D spectrum in Fig. 1 and Eq. (7), the dynamical 2D spectrum is calculated as a product of 1D dynamical lineshapes and convolved with a 2D static inhomogeneous lineshape to generate the ensemble 2D spectrum for that process. To determine the different strengths of these three contributions to the 2D spectrum, the ensemble 2D spectra for the three processes $(0 \rightarrow X, X \rightarrow 0, X \rightarrow XX)$ are added with adjustable

strengths to simulate the 2D spectrum. This approach requires that we rewrite Eq. (7) as a 2D convolution. To do this, we begin by expressing the 1D dynamical lineshapes in a form suitable for convolution. The absorption and emission spectra of single quantum dots are expected to depend not only on their size, but also upon their shape, facets, etc.(41) Dynamical absorption and emission spectra are defined as *restricted* averages of the Einstein *b*-spectra over molecules *j*' that have the same change in standard chemical potential upon photo-excitation. The group of molecules *j* with a given change in standard chemical potential $\Delta \mu_{0\to X}^{\circ}$ has a frequency of maximum absorption $\nu_{0\to X}^{dyn}$, so its dynamical absorption *b* spectrum may be written

$$b_{0\to X}^{dyn}(\nu;\nu_{0\to X}^{dyn}) = \left\langle b_{j',X\to 0}(\nu;\Delta\mu_{0\to X}^{o})\right\rangle_{j'},$$
S(3)

where the pressure and temperature dependence have been suppressed. On each side of Eq. S(3), the semicolon separates a given quantity that specifies the group of molecules *j*'. For each change in standard chemical potential, the generalized Einstein relations in Eq. (2) hold between the dynamical spectra. The dynamical absorption lineshape $g_{0\to X}^{dyn}$ is the area-normalized average of the Einstein *b* spectrum for molecules with a given change in standard chemical potential upon excitation from $0 \to X$. Approximating the shape of the *b*-spectrum as independent of $v_{0\to X}^{dyn}$ (which is justified for the small bandgap distribution probed here), we can define a normalized dynamical absorption lineshape

$$g_{0\to X}^{dyn}(\nu - \nu_{0\to X}^{dyn}) = b_{0\to X}^{dyn}(\nu; \nu_{0\to X}^{dyn}) / B_{0\to X}^{\nu}, \qquad S(4)$$

where $B_{0\to X}^{\nu} = \int_{0}^{\infty} b_{0\to X}^{dyn}(\nu) d\nu$ is the integrated Einstein *B* coefficient [which depends on the frequency units, as indicated by its right superscript – see ref. (24)]. This assumption that the dynamical absorption lineshape is independent of the frequency of maximum absorption allows both deconvolution of the linear spectra and convolution of the 2D spectra. It implies that the Stokes' shift is independent of the frequency of maximum absorption.

The change in dynamical absorption lineshape caused by $0 \rightarrow X$ excitation is a sum of three contributions given by

$$\Delta g^{dyn}(v_t - v_{0 \to X}^{dyn}) = w_{0 \to X} g^{dyn}_{0 \to X}(v_t - v_{0 \to X}^{dyn}) + w_{X \to 0} g^{dyn}_{X \to 0}(v_t - v_{0 \to X}^{dyn}) - w_{X \to XX} g^{dyn}_{X \to XX}(v_t - v_{0 \to X}^{dyn}), \quad S(5)$$

where the w_M are relative intensity strengths for the three subscripted transitions shown in Fig. 1B. These strengths are used in a linear least squares stage of the fitting procedure (see below). As a practical matter, the dynamical lineshape $g_{X\to0}^{dym}$ for excited state emission is readily area normalized, but the absorption transitions merge with transitions to higher excitons, so all three dynamical lineshapes ($g_{0\to X}^{dym}$ for reduced ground state absorption, $g_{X\to0}^{dym}$, and $g_{X\to XX}^{dym}$ for excited state emission in the emission is readily area normalized to twice the area below their 1S-1S absorption maximum. The emission lineshape suggests that errors in the fitted absorption strengths from this

approximation are on the order of 6%. With this normalization, the Einstein *b* spectra in Eq. (7) are all related to their corresponding products in Eq. S(4) and S(5) by the same proportionality constant $\beta = B_M^v / w_M$. Since the products $w_M g_M^{dyn}$ are unaffected by this lineshape normalization choice, the determination of the change in standard chemical potential does not depend on this lineshape normalization. For nanocrystals with excitonic absorption maximum at v_m , their change in standard chemical potential $\Delta \mu_{m,0\to X}^o$ equals the photon energy hv at which $w_{0\to X}g_{0\to X}^{dyn}(v-v_m) = w_{X\to 0}g_{X\to 0}^{dyn}(v-v_m)$; this criterion is equivalent to finding the photon energy at which $b_{m,X\to 0}(v, p, T) = b_{m,0\to X}(v, p, T)$ in Eq. (2)b.

The individual dynamical lineshapes in Eq. S(5) are obtained from the experimental ensemble absorption and emission lineshapes in Eq. S(2). The key relationship is convolution:(78)

$$g_{0\to X}^{ens}(\nu) = P_{static}(\nu) * g_{0\to X}^{dyn}(\nu) \equiv \int_{-\infty}^{+\infty} P_{static}(\nu_{0\to X}^{dyn}) g_{0\to X}^{dyn}(\nu - \nu_{0\to X}^{dyn}) d\nu_{0\to X}^{dyn}$$
S(6)a

$$g_{X\to 0}^{ens}(\nu) = P_{static}(\nu) * g_{X\to 0}^{dyn}(\nu)$$
 S(6)b

In fitting, the static distribution of dynamical absorption maxima is assumed to be Gaussian,

$$P_{static}(v_{0\to X}^{dyn}) = \frac{h}{\Delta_{static}\sqrt{2\pi}} \exp[-h^2(v_{0\to X}^{dyn} - v_m)^2 / 2\Delta_{static}^2], \qquad S(7)$$

with center and most probable frequency v_m and standard deviation Δ_{static} / h . An asymmetrical dynamical lineshape g can shift the ensemble absorption maximum $(v_{0\to X}^{ens})$ off the center frequency. Deconvolution can be used to obtain the dynamical absorption and emission lineshapes from the ensemble absorption and emission lineshapes. The dynamical lineshape for the excited state absorption transition from exciton to bi-exciton is obtained from the dynamical absorption lineshape by both shifting it and broadening it through convolution:

$$g_{X \to XX}^{dyn}(\nu - \nu_{0 \to X}^{dyn}) = f_{XX}(\nu) * g_{0 \to X}^{dyn}(\nu - (\nu_{0 \to X}^{dyn} - \delta_{XX}))$$
 S(6)c

where, assuming Gaussian bi-exciton broadening,

$$f_{\rm XX}(v') = \frac{h}{\Delta_{\rm XX}\sqrt{2\pi}} \exp[-h^2(v')^2 / 2\Delta_{\rm XX}^2].$$
 S(8)

The approximation of Gaussian bi-exciton broadening was based on the approximately Gaussian shape of the ensemble 1S-1S absorption.

In fitting, the ensemble absorption and emission lineshapes are deconvolved with a trial static probability distribution of absorption maxima to generate trial dynamical absorption and emission lineshapes. The 2D spectra are simulated with a diagonal convolution of dynamical 2D spectra using the same static probability distribution:

$$S_{2D}^{B}(v_{t}, v_{\tau}; T_{W}) = N_{X}(T_{W}) \int_{-\infty}^{+\infty} g_{0 \to X}^{dyn}(v_{\tau} - v_{0 \to X}^{dyn}) \Delta g^{dyn}(v_{t} - v_{0 \to X}^{dyn}) P_{static}(v_{0 \to X}^{dyn}) dv_{0 \to X}^{dyn}$$
 S(9)

where v_r is the excitation frequency, v_t is the detection frequency, and $P_{static}(v_{0\to X}^{dyn})$ is the static probability distribution of quantum dot 1S-1S dynamical absorption maxima $v_{0\to X}^{dyn}$. $N_X(T_w)$ is the waiting time dependent transient population of the exciton band X, which is assumed to be the same as the population decrease of the ground band 0. (Although they are initially equal, the transient population of X can become less than the population decrease of 0 if the population of X decays through a trap or defect.) The static distribution of dynamical absorption maxima is assumed to be Gaussian with standard deviation Δ_{static} . As shown in Fig. 1, this static distribution of bandgaps elongates the 2D spectrum along the diagonal. Compared to Eq. (7), Eq. S(9) is specialized for the assumptions that allow Eq. (7) to be expressed as a 2D convolution. Both equations assume thermal quasi-equilibrium within each band.

Chi-squared 2D fitting routine

Fitting of the 2D spectra uses only the ensemble absorption and emission lineshapes from Eq. S(2) and five physically-motivated parameters:

 Δ_{static} - the standard deviation of the inhomogeneous static bandgap distribution in Eq. S(7); δ_{XX} - the red-shift of the exciton to bi-exciton absorption transition in Eq. S(6)c; Δ_{XX} - the standard deviation for exciton to bi-exciton absorption broadening in Eq. S(8); $w_{X\to XX}$ - the strength of exciton to bi-exciton absorption in Eq. S(5); $w_{X\to 0}$ - the strength of single-exciton stimulated emission in Eq. S(5);

In reporting results, the strength of ground band 0 to exciton band X absorption is defined as $w_{0\to X} \equiv 64$ and the other strengths are reported relative to it. All five parameters are determined independently for each 2D spectrum.

The fitting procedure involves iteration of 4 processes, as summarized in Figure S3. Initially, a 3-dimensional grid is generated for the 3 parameters on which the 2D spectra have a non-linear dependence: Δ_{static} , δ_{XX} , and Δ_{XX} . At each grid point, each of the three dynamical lineshapes is normalized so that the *red half* of the 1S-1S transition has an area of 1/2. In the first process, the ensemble absorption and emission lineshapes are used together with $(\Delta_{static}, \delta_{XX}, \Delta_{XX})$ to calculate the dynamical absorption, emission, and excited state absorption lineshapes from Eq. S(6) at every 3D grid point. In the second process, at each point of the 3D grid, the three additive contributions to the 2D spectrum $(0 \rightarrow X, X \rightarrow 0, X \rightarrow XX)$ are calculated separately for each combination of the three non-linear parameters. Using Eq. S(9) and S(5), a linear least-squares fit of these three 2D contributions to the Einstein B 2D spectrum gives the three strengths and χ^2 at each 3D grid point. In the third process, calculated values of χ^2 on the 3D grid are used to estimate the 3 parameters at the χ^2 minimum. If the parameters of the estimated minimum not are enclosed within the 3D grid, the fourth process shifts the 3D grid. If 3D grid encloses the χ^2 minimum, but the parameters are not yet converged, the fourth process refines the location and resolution of the 3D grid. The above four processes are iterated until the χ^2 hypersurface is centered on and roughly parabolic around the minimum in each of the 3 nonlinear parameters.

The subroutine "Convert 1D lineshapes" in the above algorithm inverts the convolutions in Eq. S(6)a and S(6)b. Since the static lineshape width is a fit parameter, automated fitting requires accurate automated deconvolution. A van Cittert deconvolution algorithm(80) was used. For each static linewidth, the lowest van Cittert index ($k \approx 10-15$) that reproduced the FWHM obtained by analytic deconvolution of a multiple Gaussian fit to the ensemble absorption lineshape was automatically chosen (see Ref. (67) for details). This deconvolution procedure in the first subroutine allows an experimental test of the generalized Einstein relations. After presenting fit results using the deconvolution procedure, an alternative procedure based on the assumption that the generalized Einstein relations hold will be described below as an additional χ_v^2 test.

Justification for exciton to bi-exciton absorption parameters in the simulation

Figure S4 summarizes the quantitative χ_{ν}^2 justification for including excited state absorption transitions from the exciton to bi-exciton state. Without it (first column), the best fit is visually poor on the low detection frequency side of the diagonal and has a large error with a reduced chi squared of over 6. When bi-exciton broadening Δ_{XX} is added without a bi-exciton shift (second column), the fit is visually reasonable at the level of overlaid contours except upon fine inspection of the low detection frequency side and χ_{ν}^2 is significantly reduced but still exceeds 3. When a bi-exciton red-shift δ_{XX} is included without bi-exciton broadening (third column), the fit improves visually and χ_{ν}^2 is lowered significantly but the fine details of the 2D peakshape near the bandgap maximum are not captured. However, when both Δ_{XX} and δ_{XX} are included, the fit retrieves fine details on both the low detection frequency side and around the 2D maximum; χ_{ν}^2 indicates the 2D spectrum is fit within experimental error. Compared to including only the biexciton red-shift parameter, inclusion of both broadening and shift results in a significant reduction of χ^2 by ~400.

The need to include excited state absorption transitions from the exciton to bi-exciton band was qualitatively indicated by the Δ OD spectrum obtained from the measured 2D spectra by integrating over the excitation frequencies (see Figure S5). Relative to the absorption lineshape, the narrowing and blue-shift of the positive peaks in the Δ OD spectrum are signatures of excited state absorption. There has been a previous report of excited state absorption broadening in PbS QDs, however, the mechanism was suggested to be a multiphoton-excitation-induced transient Stark shift because the reported broadening depended on excitation photon energy and the excitation probability(81). With bandgap excitation and excitation probability of ~9%, we calculate less than 1 meV contribution from the previously reported multi-photon absorption broadening effect. An intrinsic exciton to bi-exciton broadening mechanism has been reported in quantum wells(42), in which the bi-exciton broadening was reported to be inversely related to the localization or the width of quantum wells. A similar intrinsic mechanism is required to explain the exciton to bi-exciton broadening found here.

Fit parameter error bars

The goodness of fit is evaluated inside a rectangular region (shown in Fig. 2 and S6) that does not have significant noise amplification. For independent data points with Gaussian noise, the closed χ^2 hypersurface that corresponds to $\chi^2_{min} + 1$ represents the boundary that contains

one standard deviation error for fit parameters.(82) For each parameter, its maximum and minimum within the closed surface give the $+\sigma$ and $-\sigma$ error bars, respectively. However, in the 2D spectrum, the sampling interval along the detection axis is very fine compared to that along the excitation axis. For each excitation frequency, neighboring points along the detection axis are highly correlated. In practice, this very fine sampling along one axis means that one could extract some number, N, of adequately sampled replicas of the 2D spectrum. The situation is thus analogous to fitting a set of N identical 2D spectra, which would increase χ^2 by a factor of N. In this circumstance, the χ^2 error bar boundary is redefined to be enclosed within $\chi^2_{min} + N$. Since data points along ω_{τ} are just adequately sampled, N is determined by the approximate number of data points along ω_t required to represent the 2D spectrum without loss of information. Therefore, we chose $N = N_t/N_{\tau}$, where N_t and N_{τ} are the number of grid points in the rectangular region for the ω_t and ω_{τ} frequency axes, respectively. The ratio usually gives $N \sim 20$ here.

Procedure for fitting 2D spectra with the generalized Einstein relations as a constraint

The 2D spectra were subsequently fit with a procedure that imposed the generalized Einstein relations as a constraint on the dynamical absorption and emission lineshapes. This procedure simultaneously fit the ensemble absorption, emission and 2D spectrum to an underlying pair of dynamical *b*-spectra. This involved a modification of the subroutine "Convert 1D lineshapes" in the fitting algorithm of Fig. S3. A flowchart for this modified subroutine is shown in Fig. S7.

The dynamical absorption and emission lineshapes that result from converged fitting of the 2D spectra need not obey the generalized Einstein relations exactly. The key to incorporating the generalized Einstein relations as a constraint is to recognize that there is a single function underlying the dynamical absorption and emission spectra, but that it is poorly determined from regions of low signal (red edge of absorption, blue edge of emission). To determine this single function, Eq. (2)b is rewritten as

$$g_{X\to0}^{dyn}(\nu) = \frac{w_{0\to X}}{w_{X\to0}} g_{0\to X}^{dyn}(\nu) \exp[-(h\nu - \Delta \mu_{0\to X}^{\circ})/kT].$$
 S(10)

Starting from the input data from the prior 2D fitting iteration, there are four initialization steps before nonlinear least squares fitting is used to find the best fit dynamical absorption lineshape. First, the standard chemical potential change $\Delta \mu_{0\to X}^{o}$ is found from Eq. S(10) as the photon energy hv at which $w_{X\to 0}g_{X\to 0}^{dyn}(v)\Big|_{\Delta \mu_{0\to X}^{o}/h} = w_{0\to X}g_{0\to X}^{dyn}(v)\Big|_{\Delta \mu_{0\to X}^{o}/h}$. Second, the best fit dynamical absorption and emission lineshapes from the prior 2D fitting iteration are both used to generate an initial guess for the dynamical absorption lineshape

$$\gamma_{0\to X}^{dyn}(\nu) = [1 - \operatorname{switch}(\nu)] \frac{w_{X\to 0}}{w_{0\to X}} g_{X\to 0}^{dyn}(\nu) \exp[+(h\nu - \Delta\mu_{0\to X}^{\circ})/kT] + \operatorname{switch}(\nu) g_{0\to X}^{dyn}(\nu),$$

$$S(11)$$

where switch is a function that smoothly transitions from 0 to 1 between the emission and absorption maxima. Third, the initial guess for corresponding generalized Einstein relation constrained dynamical emission lineshape $\gamma_{X\to 0}^{dyn}(\nu)$ is calculated by inserting $\gamma_{0\to X}^{dyn}(\nu)$ from the

left hand side of Eq. S(11) into the right hand side of Eq. S(10). Fourth, the initial guesses for the dynamical absorption and emission lineshapes are then convolved with the prior best fit static bandgap distribution Δ_{static}^{o} using Eq. S(6)ab to obtain corresponding initial guesses for the ensemble absorption and emission lineshapes.

Starting from these 4 guesses, a weighted sum of the 4 rms errors for the dynamical absorption and emission (each compared to the best fit lineshape from the prior iteration of the 2D fit) and ensemble absorption and emission (each compared to the experimental lineshape) is then minimized by independently varying the dynamical absorption lineshape at every point on a frequency grid. Absorption lineshape errors were weighted 9x more than emission lineshape errors. To ensure a non-negative dynamical absorption lineshape, the parameters varied are the square roots of the dynamical absorption lineshape. This nonlinear least squares fit (NLLSQ fit to dynamical and ensemble 1D spectra) uses Powell's method. The converged result is a set of 4 lineshapes $\gamma(\nu)$ in which the dynamical absorption and emission *b*-spectra obey the generalized Einstein relation exactly and the ensemble lineshapes are obtained from the dynamical lineshapes by numerically exact convolution. In this constrained fit, the residuals are now spread over both the 2D spectra and the ensemble 1D spectra. In contrast, the unconstrained procedure that uses only van Cittert deconvolution for the "Convert 1D spectra" subroutine in Fig. S3 effectively forces an exact fit to the 1D spectra. Both procedures assume a Gaussian static broadening.

The pair of converged ensemble absorption and emission lineshapes are then used in place of the experimental ensemble lineshapes for van Cittert deconvolution with the trial inhomogeneities Δ_{static} on the 3D grid. When implementing the fit with this more involved subroutine, the initial parameters were taken from the prior best fit in which this subroutine consisted only of van Cittert deconvolution of the experimental ensemble lineshapes. A range of inhomogeneities on the 3D grid somewhat smaller than the error bars was sufficient to verify the local minimum was hardly unchanged and converged within three iterations of 2D fitting, which gave $\chi^2_{\nu} = 1.05$ (using the same number of degrees of freedom as the unconstrained fit) and parameters that agreed within the error bars from the unconstrained fit.

The converged dynamical and ensemble lineshapes from the NLLSQ fit are shown in Fig. 3A. Using converged dynamical lineshapes taken directly from the NLLSQ fit, which exactly obey the generalized Einstein relations, to fit the 2D spectra also gave $\chi_{\nu}^2 = 1.05$ and no significant change in fit parameters. The slight errors in the fitting the experimental ensemble emission lineshape in Fig. 3B are probably due to slight deviations from Gaussian static inhomogeneous broadening of the type demonstrated in. ref. (19). The generalized Einstein relations hold within the accuracy of the 2D spectra for this sample.

Data extraction for Figure S8

Tables S2 and S3 show absorption maxima, linewidths, and Stokes' shifts extracted from the graph in Figure 1b and directly from the spectra in Figure 2d of ref. (17). In both tables, data in roman font were obtained by magnifying the figures and making measurements with a ruler; this had a precision of about 2 meV. Comparison to numbers stated in the paper suggests errors of ~10 meV. Each of the two PbS quantum dot samples in Figure 2d has extracted parameters that match a sample in Figure 1b within this error estimate. In Table S2, dynamical emission FWHM at 300K are calculated from the generalized Einstein relation using Eq. (8) for Gaussian spectra by multiplying the ensemble Stokes' shift $(2\lambda)/e$ in the third column by $(kT/e) \sim 25.85$ meV, taking the square root to find the standard deviation of the lineshape Δ/e , and then multiplying by $2\sqrt{(2\ln 2)} \sim 2.3548$ to find the FWHM. Data were also extracted from Fig. 2d using WebPlotDigitizer.(83) This extraction allowed conversion to Einstein B coefficient lineshapes in order to assess the impact of the lineshape conversion. If the absorption spectra shown are OD(v) and the photoluminescence spectra shown are F(v), converting to Einstein *B* coefficient lineshapes could increase the Stokes' shift by up to 5 meV, with negligible effect on the linewidths. The numbers reported in Table S3 are those obtained from Figure 2d without this lineshape conversion.

Supplementary Text

Derivation of the extended Einstein b relationships for inhomogeneous Gaussian spectra

The stimulated emission spectrum from the generalized Einstein relation is red-shifted from the absorption spectrum. The red-shift is called the Stokes' shift. More precise consequences of the generalized Einstein relation, Eq. (2)b, for the Stokes' shift and integrated Einstein *B* coefficients can be illustrated by considering a homogeneous Gaussian absorption lineshape:

$$b_{0\to X}^{dyn}(\nu;\nu_{0\to X}^{dyn}) = B_{0\to X}^{\nu} \cdot \frac{h}{\Delta_{dyn}\sqrt{2\pi}} \exp\left[-h^2\left(\nu - \nu_{0\to X}^{dyn}\right)^2 / 2\Delta_{dyn}^2\right]$$
 S(12)a

where $v_{0\to X}^{dyn}$ is the frequency of maximum absorption, Δ_{dyn} / h is the standard deviation that quantifies the Gaussian linewidth (the subscript indicates the dynamical origin of a homogeneous lineshape), and $B_{0\to X}^{\nu} = \int_{0}^{\infty} b_{0\to X}^{dyn}(\nu) d\nu$ is the integrated Einstein *B* coefficient. The superscript on *B* indicates units appropriate for cyclic frequencies (Hz).(24) The pressure and temperature dependence of both *b* and *B* have been suppressed. The generalized Einstein relation Eq. (2)b gives the stimulated emission *b*-spectrum

$$b_{X\to 0}^{dyn}(\nu;\nu_{0\to X}^{dyn}) = B_{0\to X}^{\nu} \cdot \frac{h}{\Delta_{dyn}\sqrt{2\pi}} \exp\left[-\frac{h^2(\nu-\nu_{0\to X}^{dyn})^2}{2\Delta_{dyn}^2} - \frac{(h\nu-\Delta\mu_{0\to X}^{\circ})}{kT}\right].$$

The frequency of maximum emission and integrated Einstein B coefficient for stimulated emission can be found by completing the square inside the exponential on the second line. This gives the frequency of maximum emission

$$hv_{X\to 0}^{dyn} = hv_{0\to X}^{dyn} - \Delta_{dyn}^2/kT ,$$

so the Stokes' shift for homogeneous Gaussian absorption and emission transitions is

$$2\lambda = hv_{0\to X}^{dyn} - hv_{X\to 0}^{dyn} = \Delta_{dyn}^2/k_B T$$
 S(13)

which may be rewritten as

$$\Delta_{dvn}^2 = 2\lambda kT \ . \tag{8}$$

Eq. (8) appears in molecular electronic spectra(84) and electron transfer,(46) but is a consequence of the generalized Einstein relations for any homogeneous Gaussian spectrum. For a homogeneous Gaussian absorption lineshape, the stimulated emission lineshape is a Gaussian with the same width, but a red-shifted emission maximum; fundamentally, this Stokes' shift determines the width of both the absorption and stimulated emission spectra. We now address the relationship between integrated Einstein *B* coefficients. We define $v_0^{dyn} = (v_{0\to X}^{dyn} + v_{X\to 0}^{dyn})/2$ so that it lies halfway between the dynamical absorption and emission maxima. This gives the stimulated emission *b*-spectrum as

$$b_{X\to0}^{dyn}(\nu;\nu_{0\to X}^{dyn}) = B_{0\to X}^{\nu} \cdot \frac{h}{\Delta_{dyn}\sqrt{2\pi}} \exp\left[-\frac{h^2(\nu-\nu_{X\to0}^{dyn})^2}{2\Delta_{dyn}^2}\right] \exp\left[-\frac{h\nu_0^{dyn}-\Delta\mu_{0\to X}^0}{kT}\right]$$
 S(12)b

so that the integrated Einstein B coefficient for stimulated emission is

$$B_{X\to0}^{\nu} = \int_{0}^{\infty} b_{X\to0}^{dyn}(\nu) d\nu$$
$$= B_{0\to X}^{\nu} \exp\left[-\frac{h\nu_{0}^{dyn} - \Delta\mu_{0\to X}^{o}}{kT}\right].$$
 S(14)

Any given quantum dot sample will have a static distribution of changes in standard chemical potential upon photo-excitation; the ensemble lineshapes are averaged over this static distribution and will not obey the generalized Einstein relations. In the treatment below, each pair of dynamical lineshapes will be characterized by its frequency of maximum absorption as in Eq. S(3). Quantum dot absorption spectra usually have an overlap between the blue edge of the lowest exciton transition (1S-1S) and transitions to higher exciton states so that there is a nonzero minimum on the high energy side of the first maximum (see Fig. 3). Provided the lowest exciton transition is wide enough and the temperature is low enough, this blue edge is heavily attenuated by the $\exp(-hv/kT)$ factor and has almost no effect on the emission spectrum predicted by the generalized Einstein relations. If the red edge of the dynamical absorption spectrum has the Gaussian lineshape of Eq. S(12)a, then the dynamical lineshapes defined through Eq. S(4) with the static bandgap distribution centered at $v_m = v_{0 \to X}^{ens}$ in Eq. S(7) gives the ensemble spectra

$$\left\langle b_{0\to X}(\nu) \right\rangle = B_{0\to X}^{\nu} \cdot \frac{h}{\Delta_{ens} \sqrt{2\pi}} \exp\left[-h^2 \left(\nu - \nu_{0\to X}^{ens}\right)^2 / 2\Delta_{ens}^2\right]$$
 S(15)a

$$\left\langle b_{\mathrm{X}\to0}(\nu)\right\rangle = B_{\mathrm{X}\to0}^{\nu} \cdot \frac{h}{\Delta_{ens}\sqrt{2\pi}} \exp\left[-h^2\left(\nu - \left(\nu_{0\to\mathrm{X}}^{ens} - 2\lambda/h\right)\right)^2 / 2\Delta_{ens}^2\right]$$
 S(15)b

where the variance of the ensemble-averaged spectra has increased to

$$\Delta_{ens}^2 = \Delta_{dyn}^2 + \Delta_{static}^2 \,, \tag{9}$$

yet the Stokes' shift is still given by Eq. (8). The ensemble Stokes' shift can be distorted if the bandgap distribution is wide enough to sample a nonlinear variation of the Stokes' shift with bandgap. The ensemble Stokes' shift can also be different from the dynamical Stokes' shift if the bandgap distribution is wide enough to shift higher exciton transitions on top of the lowest

exciton absorption maximum. By applying the generalized Einstein relations directly to either ensemble spectrum as a test for static inhomogeneity, the Stokes' shift would be over-estimated as $2\Lambda = \Delta_{ens}^2 / kT$. If the red edge of the ensemble absorption spectrum is Gaussian and the ensemble emission spectrum is Gaussian with the same width, this justifies using the ensemble Stokes' shift 2λ to determine the variance of the dynamical linewidth Δ_{dyn}^2 through Eq. (8).

This lack of distortion for narrow bandgap distributions justifies use of the ensemble Stokes' shift to determine the dynamical linewidth. When the ensemble lineshape is near Gaussian (which is not the case for the PbS quantum dots studied here), these circumstances allow algebraic estimates of static and dynamical linewidths to be made from ensemble absorption and emission spectra alone, without numerical deconvolution or specialized nonlinear optical spectroscopy apparatus.

For the PbS quantum dot samples studied by 2D spectroscopy here, the dynamical lineshapes are not Gaussian (see Fig. 3, which also shows that the ensemble absorption and emission lineshapes have different widths). For the part-covered PbS quantum dots, the measured Stokes' shift is $(2\lambda)/e = 88 \text{ meV}$, for which Eq. (8) gives $\Delta_{dyn}/e = 48 \text{ meV}$ at 298 K. The ensemble emission $(FWHM_{ens})/e$ is 119 meV, corresponding to a Gaussian standard deviation of $\Delta_{ens}/e = 52 \text{ meV}$. Eq. (9) then gives $\Delta_{static}/e = 20 \text{ meV}$, a significant underestimate. One can avoid assuming that the dynamical or ensemble lineshapes are Gaussian by mis-applying the generalized Einstein relations to calculate an ensemble emission spectrum directly from the ensemble absorption spectrum (as in Fig. 3C) to determine $(2\Lambda)/e = 112 \text{ meV}$. Calculating $\Delta_{static}^2 = 2(\Lambda - \lambda)kT$ gives $\Delta_{static}/e = 24 \text{ meV}$, which agrees with the 2D estimate within error. The broader applicability of this approach remains to be investigated.

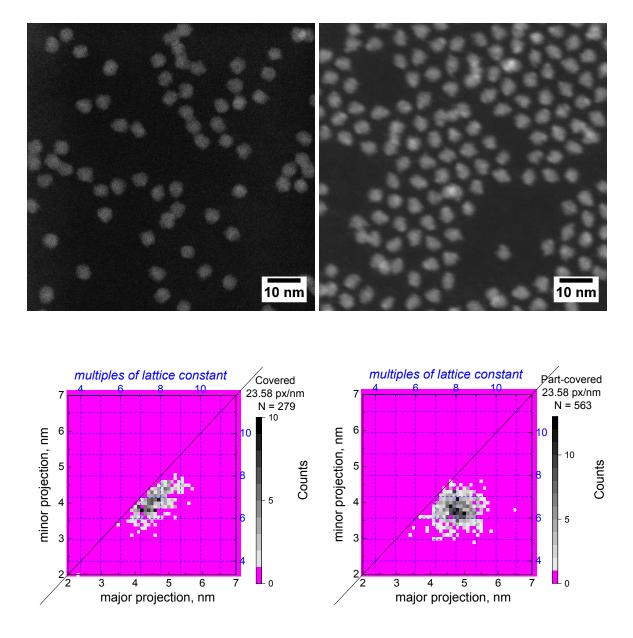


Fig. S1. Electron microscopy images of PbS nanocrystals and histograms of their majorminor axis projections. ADF-STEM images of (upper left) ligand-covered PbS nanocrystals and (upper right) part-covered PbS nanocrystals. Images were collected on FEI Titan at 300 kV accelerating voltage (instrumental spatial resolution 0.14 nm) by a trained technician at the CAMCOR materials characterization facility at the University of Oregon in Eugene, Oregon. Images have contrast of ~0.5, and a signal to noise ratio of ~5:1. A scale bar is drawn on the lower right corner of each image. 2D histogram of major versus minor projections from threshold-fitted ellipses on ADF-STEM images (image spatial resolution 0.14 nm) for covered (lower left) and part-covered (lower right) PbS nanocrystal samples. The magenta background for each histogram represents counts of zero particles, and the grayscale bar is used to represent the number of counts at each 2D histogram. Each histogram is normalized to the highest count in its distribution. The pixel pitch for each ADF-STEM image is indicated on the right-hand corner of each histogram.

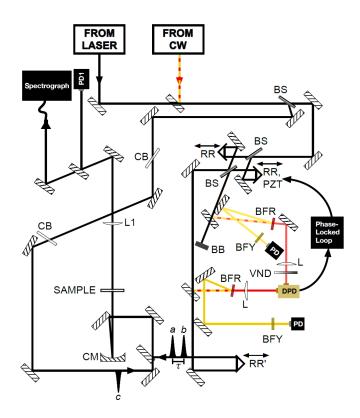


Fig. S2. Partially collinear 2D spectrometer. Pulses *a* and *b*, separated by delay τ , exit the Mach-Zehnder interferometer's bright output. The signal and pulse *c* are fiber coupled to an IR spectrograph for interference detection. Each of pulses *a*, *b*, and *c* passes through two metallic beam splitters at oppositely signed Brewster's angles for matched dispersion and spatial compensation. Continuous wave (CW) lasers (beam paths marked with red and yellow lines) are used to stabilize the delay τ and travel a path below the pulsed laser beam. BS: inconel-coated beam splitter, CB: compensating window, CM: curved silver mirror, f = 125 mm, RR: dihedral retroreflector, RR': trihedral retroreflector, VND: variable neutral density filter, BFR: red bandpass filter (632.8 ± 5 nm), BFY: yellow bandpass filter (600 ± 5 nm), PD: photodiode for yellow CW, PD1: photodiode for pump-pulse pair interference, DPD: differential photodiode for active stabilization, PZT: piezoelectric transducer, BB: beam block, L1: CaF2 plano-convex lens for recollimating the beams, f = 125 mm, L: plano-convex lens, f = 50 mm. Protected silver mirrors are unlabeled.

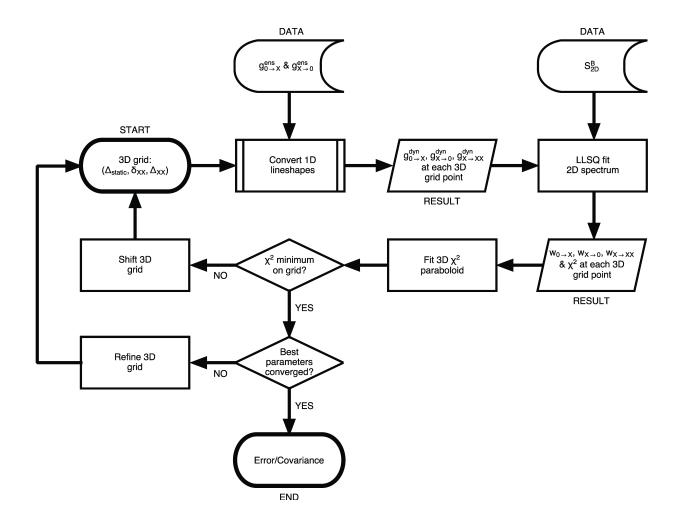
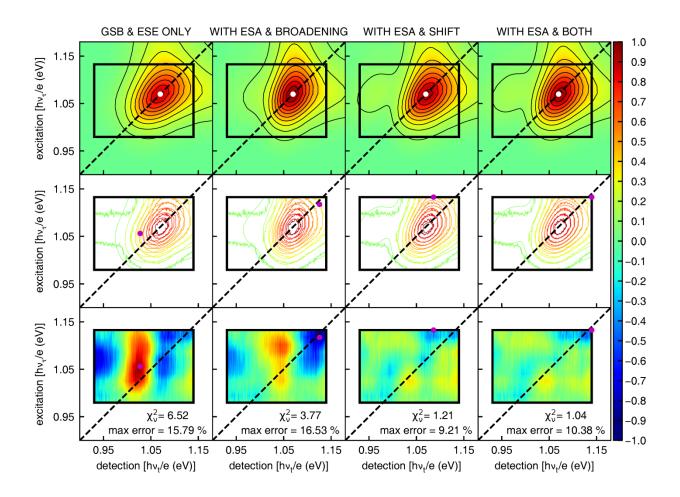
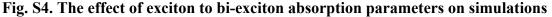


Fig. S3. Flow chart of the chi-squared fitting procedure

The flowchart describes the fitting procedure, which involves cycles of 4 steps. Before starting each cycle, a 3D grid of static bandgap inhomogeneities Δ_{static} , exciton to bi-exciton red-shifts δ_{xx} , and exciton to bi-exciton broadenings Δ_{xx} is generated. In the first subroutine, 1D ensemble lineshapes are converted to 1D dynamical lineshapes for each Δ_{static} . In the second process, the three contribution to the 2D spectrum are calculated for each point on the 3D grid and linear least squares fit to the experimental 2D spectra by optimizing their relative strengths $(w_{0\to x}, w_{X\to 0}, w_{X\to XX})$. χ^2 is calculated for each fitted 2D spectrum, resulting in 3D grid of χ^2 values. In the third process, a 3D paraboloid is fit to the χ^2 hypersurface to estimate where the minimum χ^2 lies. The fourth process either shifts the grid (if the minimum χ^2 is not within the grid) or refines the grid (if the best parameter estimated from the 3D paraboloid fit is not centered on the grid or the hypersurface is not yet accurately represented by a 3D paraboloid over the grid range). Once the 3D grid and fit have converged, the fit parameter errors are obtained from the boundary corresponding to $\chi^2_{\min} + N_t/N_\tau$.





The effect of exciton to bi-exciton excited state absorption (ESA) parameters on simulations is shown. In each column, the top row shows the simulation, the middle row shows simulation overlaid with the experimental data, and the bottom row shows their difference (simulation – experiment) multiplied by 10. Contours are drawn from 10% at 10% interval. When only depletion of ground to exciton absorption (ground state bleaching - GSB) and exciton to ground band stimulated emission (excited state emission – ESE) are included (first column), the fit does not agree within error, and the maximum error (purple dot) occurs within the fit region. With ESA added but only the broadening parameter (second column), χ^2_{ν} is significantly reduced but the fit still does not agree within experimental error. The inclusion of ESA with a red-shift but without broadening (third column) nearly fits the data to within error, but fine details of the fit are not captured near the maximum of the 2D signal. Addition of both ESA effects (right column) fits the data capturing the fine details near the part-covered sample's bandgap.

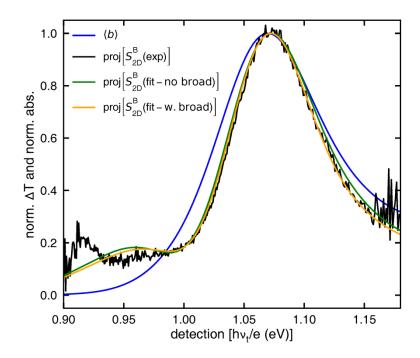


Fig. S5. Comparison of absorption lineshape with differential transmittance spectrum Normalized differential transmittance spectra ΔT are obtained from projections of the Einstein B coefficient 2D spectra onto the detection axis at waiting time $T_W = 1$ ps. All 3 projections are obtained by integrating the 2D spectrum over the excitation frequency from $hv_r/e = 0.91$ to 1.14 eV (which extends beyond the fitting box in Fig. 2). The ensemble linear absorption Einstein *B* coefficient lineshape of the part-covered PbS QDs (blue) is overlaid and normalized at the peak. The experimental projection (black) is narrower and shifted to higher frequency. Since emission can only broaden the ΔT spectrum, the narrower width of the experimental ΔT spectrum compared to $\langle b \rangle$ indicates that excited state absorption is necessary to reproduce the shift and narrowing of the ΔT lineshape. The peak normalized ΔT spectrum from the simulation that includes shifted excited state absorption without broadening cannot quite capture the experimental ΔT width. The peak normalized ΔT spectrum from the simulation (orange) that includes shifted and broadened excited state absorption captures this shift and narrowing. This qualitative visual demonstration supports the reduced chi-squared analysis (column 3 vs. column 4 in Fig. S4) indicating that the excited state absorption signal is broadened.

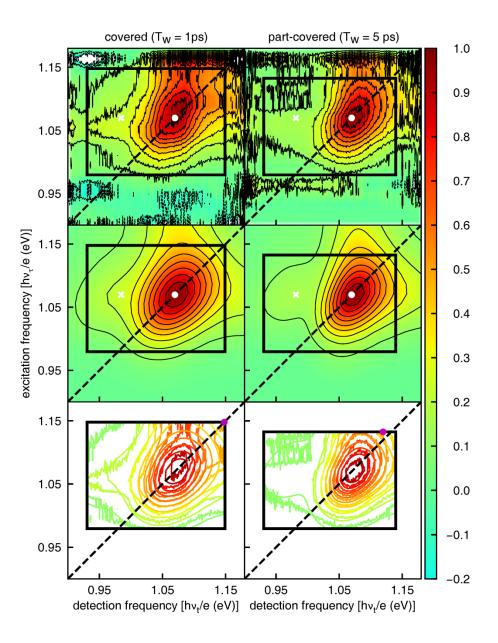


Fig. S6. 2D spectra of covered PbS quantum dots at $T_w=1$ ps, part-covered PbS quantum dots at $T_w=5$ ps and their best fits

Contour levels are drawn from -20% to 100% in 10% increments. (Top) 2D relaxation spectra after division to correct for pulse spectra and propagation effects. Noise is 3.90% and 2.44% inside the box. (Middle) Fit using absorption, stimulated emission, and shifted, broadened excited state absorption ($\chi_{\nu}^2 = 1.99$ and 2.83). A white dot on each experimental and fit 2D spectrum indicates the 1S–1S peak maximum from the linear absorption lineshape, and a white cross marks the 2D coordinates for maximum 1S-1S stimulated emission. (Bottom) Experiment and fit overlaid at 10% contours. The RMS error of the fit is 7.90% and 6.84%. The maximum error of the fit is 43% and 17%, which occurs at the position indicated by the magenta dot (upper right corner of the box). Best fit parameters are given in Table 1.

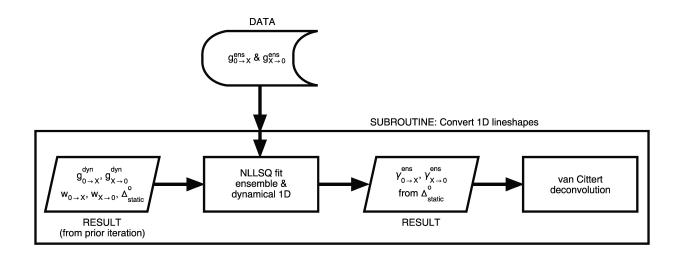


Fig. S7. Converting 1D lineshapes with generalized Einstein relations as a constraint

This constrained process inserts additional steps before the automated van Cittert deconvolution used to calculate the dynamical spectra for each 3D grid point. The single optimum set of dynamical 1D lineshapes, strengths, and static inhomogeneity Δ_{static}° from the prior 2D fitting iteration (left result) are used to impose the generalized Einstein relations as a constraint. The dynamical emission spectrum is calculated from the dynamical absorption spectrum using Eq. S(10) and the ensemble 1D spectra are calculated from the dynamical spectra using Eq. S(6)ab for the optimum static inhomogeneity Δ_{static}° . The non-linear least squares (NLLSQ) fit process adjusts the underlying dynamical absorption lineshape $\gamma_{0\to X}^{dyn}(\nu)$ to minimize a weighted sum of the rms errors for the input dynamical absorption and emission lineshapes (from the prior 2D fitting iteration) and the ensemble absorption and emission lineshapes (from experiment). The best fit 1D ensemble lineshapes $\gamma_{0\to X}^{ens}(\nu)$ and $\gamma_{X\to 0}^{ens}(\nu)$ for Δ_{static}° (right parallelogram) are then used in place of the experimental ensemble lineshapes for van Cittert deconvolution at each 3D grid point. The resulting set of three dynamical lineshapes is then used as an input to the "LLSQ fit 2D spectrum" process of Fig. S3.

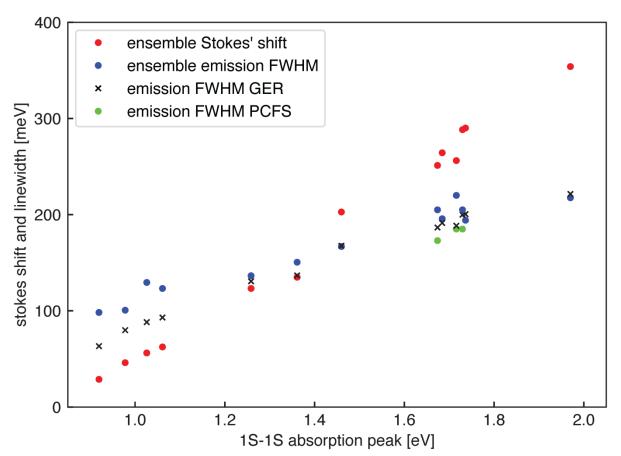


Fig. S8. Comparison of generalized Einstein relation prediction to photon-correlation Fourier spectroscopy

Ensemble Stokes' shift (red dots), ensemble emission FWHM (blue dots), and photon-correlation Fourier spectroscopy determined emission FWHM (green dots) for PbS nanocrystals (prepared by different syntheses from either used here) were extracted from Figure 1(b) and 2(d) of ref. (17) with estimated graphical extraction errors of 3 meV or less. Dynamical emission FWHM (black cross) values are calculated from the ensemble Stokes' shift using Eq. (7) of the main text. For all samples in which the dynamical emission FWHM was measured by photon correlation Fourier spectroscopy, the emission FWHM calculated from the generalized Einstein relation lies between the ensemble and dynamical emission FWHM. Since ref. (17) concludes these PbS nanocrystals are homogeneously broadened, this is taken as agreement within experimental error (~10 meV). All PbS nanocrystals with 1S-1S absorption peaks above 1.2 eV show agreement between the measured ensemble emission FWHM and the FWHM calculated from the homogeneous generalized Einstein relation. This indicates that the ensemble linewidth for the smaller nanocrystals is dominated by dynamical broadening, not static inhomogeneous broadening due to their size distribution. In ref. (17), photon correlation spectra were not reported for four larger PbS quantum dot sizes with bandgaps below 1.2 eV. For these larger dots, the ensemble Stokes' shift and generalized Einstein relations predict narrower dynamical emission FWHM than the reported ensemble emission FWHM. This indicates significant inhomogeneous broadening of the ensemble spectra from the static size distribution.

	area equivalent circle		ellipse projections			
	diameter	σ (diameter)	major	σ (major)	minor	σ (minor)
covered [†]	4.29(11)	0.32(8)	4.58(13)	0.38(8)	4.01(9)	0.33(9)
part-covered*	4.27(9)	0.23(2)	4.81(9)	0.37(4)	3.80(12)	0.28(1)

Table S1. Particle size analyses from ADF-STEM (threshold) [dimensions in nm (error bars)].*563 particles (5 images, 0.0424 nm/pixel); [†]279 particles (10 images, 0.0424 nm/pixel).

Absorption Peak (eV)	Ensemble Emission FWHM (meV)	Ensemble Stokes' Shift (meV)	Dynamical emission FWHM from GER (meV)
0.921	97	28	63
1.026	130	55	89
1.061	123	61	93
1.258	134	122	132
1.360	147	134	139
1.459	165	200	169
1.684	194	264	195
1.736	195	289	204
1.968	216	354	225

Table S2. Ensemble absorption maximum, ensemble emission FWHM, and Stokes' shift extracted from Figure 1b of Ref. (17) for PbS quantum dots. Parameters were not extracted for the sample with the second lowest bandgap, which has an obscured ensemble emission FWHM in Figure 1b. Dynamical emission FWHM are calculated from the Stokes' shift and generalized Einstein relation for Gaussian spectra using a temperature of 300K.

Sample Type	Absorption Peak (eV)	Ensemble Emission FWHM (meV)	PCFS Emission FWHM (meV)	Ensemble Stokes' Shift (meV)
PbS (2)	1.676	205 vs. 192	173 vs.173	253
PbS/CdS (2)	1.721	220 vs. 209	185 vs. 184	259
PbS (1)	1.733	205 vs. 193	185 vs. 184	292

Table S3. Ensemble and dynamical linewidth parameters extracted from ensemble and photoncorrelation Fourier spectroscopy (PCFS) spectra of PbS and PbS/CdS core/shell quantum dots in Figure 2d of Ref. (17). The numerical emission FWHM in italics are stated in Figure 2d and are included for reference. Parameters are not extracted for the two samples in which the absorption spectrum does not show a maximum.

Data Set S1. Data files for 1D absorption spectra, 1D emission spectra, and 2D spectra. (absorbance spectrum.txt) 1D ensemble decadic absorbance or optical density spectra for both covered and part-covered PbS quantum dot samples as a function of wavelength in air at atmospheric pressure in Boulder. The first column is air wavelength in nm and the second column is optical density (dimensionless). For the part-covered sample, the wavelength in air was transformed to wavelength in vacuum before plotting the absorbance spectrum shown in Fig. 4A. (abs lineshape.txt) The ensemble absorption lineshape calculated from Eq. (14). The first column contains the photon energy in eV and the second column contains the peak normalized lineshape; the lineshapes are shown in Fig. 4C for both samples. (pl spectrum.txt) 1D ensemble photoluminescence spectra as a function of wavelength in air at atmospheric pressure in Boulder, corrected for the inner filter effect and wavelength dependent spectrograph sensitivity to photons/nm. The first column is air wavelength in nm and the second column is corrected photoluminescence (proportional to photons/nm). For the part-covered sample, the wavelength in air was transformed to wavelength in vacuum before plotting the corrected photoluminescence spectrum in Fig. 4D. (ems lineshape.txt) The ensemble stimulated emission lineshapes calculated from Eq. (16)-(18). The first column contains the photon energy in eV and the second column contains the peak normalized lineshape; the lineshapes are shown in Fig. 4F for both samples. (GER lineshape.txt) The dynamical absorption and stimulated emission lineshapes that result from using the generalized Einstein relation as a constraint when fitting the 1D and 2D spectra for the part-covered sample at T w=1 ps. These dynamical lineshapes are shown in Fig. 3A. (pl spectrum to lineshape factors.txt) the first column gives the multiplier for transformation of wavelengths in air at atmospheric pressure in Boulder to wavelengths in vacuum(76); the second column is proportional to the multiplier for transformation [combining Eq. (16) and Eq. (18)] of the corrected photoluminescence spectrum (photons/nm) to Einstein B coefficient stimulated emission lineshape b(v). (TWOD exp.txt and **TWOD** sim.txt) Experimental and simulated 2D spectra in the Einstein *B* representation. These 2D spectra are shown in Fig. 2 and Fig. S6. (README.txt) explains file formats and provides the directory structure used to distinguish spectra for different samples and different 2D waiting times.

REFERENCES AND NOTES

- R. T. Ross, M. Calvin, Thermodynamics of light emission and free-energy storage in photosynthesis. *Biophys. J.* 7, 595–614 (1967).
- 2. R. E. Blankenship, Molecular Mechanisms of Photosynthesis (Blackwell Science Ltd., 2002).
- W. Shockley, H. J. Queisser, Detailed balance limit of efficiency of *p-n* junction solar cells. *J. Appl. Phys.* 32, 510–519 (1961).
- 4. P. Würfel, U. Würfel, *Physics of Solar Cells* (Wiley-VCH, ed. 3, 2016).
- 5. A. Einstein, in Sources of Quantum Mechanics, B. L. van der Waerden, Ed. (Dover, 1967), pp. 63-77.
- B. I. Stepanov, A universal relation between the absorption and luminescence spectra of complex molecules. *Sov. Phys.-Dokl.* 2, 81–84 (1957).
- S. J. Strickler, R. A. Berg, Relationship between absorption intensity and fluorescence lifetime of molecules. *J. Chem. Phys.* 37, 814–822 (1962).
- E. McCumber, Einstein relations connecting broadband emission and absorption spectra. *Phys. Rev.* 136, A954–A957 (1964).
- 9. R. T. Ross, Radiative lifetime and thermodynamic potential of excited states. *Photochem. Photobiol.* 21, 401–406 (1975).
- R. L. Van Metter, R. S. Knox, On the relation between absorption and emission spectra of molecules in solution. *Chem. Phys.* 12, 333–340 (1976).
- Y. B. Band, D. F. Heller, Relationships between the absorption and emission of light in multilevel systems. *Phys. Rev. A* 38, 1885–1895 (1988).
- S. Park, T. Joo, Diffractive optics based three-pulse photon echo peak shift studies of spectral diffusion in polar liquids: Evidence for long lived frequency correlations. *J. Chem. Phys.* 131, 164508 (2009).

- A. Hagfeldt, M. Graetzel, Light-induced redox reactions in nanocrystalline systems. *Chem. Rev.* 95, 49–68 (1995).
- 14. B. H. Kim, J. H. Heo, S. Kim, C. F. Reboul, H. Chun, D. Kang, H. Bae, H. Hyun, J. Lim, H. Lee, B. Han, T. Hyeon, A. P. A. Alivisatos, P. Ercius, H. Elmlund, J. P. Park, Critical differences in 3D atomic structure of individual ligand-protected nanocrystals in solution. *Science* **368**, 60–67 (2020).
- M. C. Weidman, M. E. Beck, R. S. Hoffman, F. Prins, W. A. Tisdale, Monodisperse, air-stable PbS nanocrystals via precursor stoichiometry control, *ACS Nano* 8, 6363–6371 (2014).
- M. P. Hendricks, M. P. Campos, G. Cleveland, I. Jen-La Plante, J. S. Owen, A tunable library of substituted thiourea precursors to metal sulfide nanocrystals. *Science* 348, 1226–1230. (2015).
- J. R. Caram, S. N. Bertram, H. Utzat, W. R. Hess, J. A. Carr, T. S. Bischof, A. P. Beyler, M. W. B. Wilson, M. G. Bawendi, PbS nanocrystal emission is governed by multiple emissive states. *Nano Lett.* 16, 6070–6077 (2016).
- Y. Dong, T. Qiao, D. Kim, D. Parobek, D. Rossi, D. H. Son, Precise control of quantum confinement in cesium lead halide perovskite quantum dots via thermodynamic equilibrium. *Nano Lett.* 18, 3716– 3722 (2018).
- S. D. Park, D. Baranov, J. Ryu, B. Cho, A. Halder, S. Seifert, S. Vajda, D. M. Jonas, Bandgap inhomogeneity of a PbSe quantum dot ensemble from two-dimensional spectroscopy and comparison to size inhomogeneity from electron microscopy. *Nano Lett.* 17, 762–771 (2017).
- 20. J. Cui, A. P. Beyler, I. Coropceanu, L. Cleary, T. R. Avila, Y. Chen, J. M. Cordero, S. L. Heathcote, D. K. Harris, O. Chen, J. Cao, M. G. Bawendi, Evolution of the single-nanocrystal photoluminescence linewidth with size and shell: Implications for exciton-phonon coupling and the optimization of spectral linewidths. *Nano Lett.* 16, 289–296 (2016).
- J. D. Hybl, A. W. Albrecht, S. M. Gallagher Faeder, D. M. Jonas, Two-dimensional electronic spectroscopy. *Chem. Phys. Lett.* 297, 307–313 (1998).

- S. Mukamel, Multidimensional femtosecond correlation spectroscopies of electronic and vibrational excitations. *Annu. Rev. Phys. Chem.* 51, 691–729 (2000).
- D. M. Jonas, Two-dimensional femtosecond spectroscopy. *Annu. Rev. Phys. Chem.* 54, 425–463 (2003).
- R. C. Hilborn, Einstein coefficients, cross sections, f values, dipole moments, and all that. *Am. J. Phys.* 50, 982–986 (1982).
- 25. V. Balevičius Jr., T. Wei, D. Di Tommaso, D. Abramavicius, J. Hauer, T. Polívka, C. D. P. Duffy, The full dynamics of energy relaxation in large organic molecules: From photo-excitation to solvent heating. *Chem. Sci.* **10**, 4792–4804 (2019).
- 26. J. G. Kirkwood, I. Oppenheim, Chemical Thermodynamics (McGraw-Hill, 1961).
- 27. C. Cohen-Tannoudji, J. Dupont-Roc, G. Grynberg, *Atom-Photon Interactions: Basic Processes and Applications* (Wiley-Interscience, 1992).
- M. Planck, *The Theory of Heat Radiation* (Dover Press, ed. Translation of the Second (1912) German Edition, 1959).
- K. Bian, B. T. Richards, H. Yang, W. Bassett, F. W. Wise, Z. Wang, T. Hanrath, Optical properties of PbS nanocrystal quantum dots at ambient and elevated pressure. *Phys. Chem. Chem. Phys.* 16, 8515–8520 (2014).
- 30. C. Kittel, H. Kroemer, Thermal Physics (W. H. Freeman, ed. 2, 1980).
- 31. D. Baranov, R. J. Hill, J. Ryu, S. D. Park, A. Huerta-Viga, A. R. Carollo, D. M. Jonas, Interferometrically stable, enclosed, spinning sample cell for spectroscopic experiments on airsensitive samples. *Rev. Sci. Instrum.* 88, 014101 (2017).
- W. Qin, P. Guyot-Sionnest, Evidence for the role of holes in blinking: Negative and oxidized CdSe/CdS dots. ACS Nano 6, 9125–9132 (2012).

- 33. F. Fan, O. Voznyy, R. P. Sabatini, K. T. Bicanic, M. M. Adachi, J. R. McBride, K. R. Reid, Y.-S. Park, X. Li, A. Jain, R. Quintero-Bermudez, M. Saravanapavanantham, M. Liu, M. Korkusinski, P. Hawrylak, V. I. Klimov, S. J. Rosenthal, S. Hoogland, E. H. Sargent, Continuous-wave lasing in colloidal quantum dot solids enabled by facet-selective epitaxy. *Nature* 544, 75–79 (2017).
- J. C. Johnson, K. A. Gerth, Q. Song, J. E. Murphy, A. J. Nozik, G. D. Scholes, Ultrafast exciton fine structure relaxation dynamics in lead chalcogenide nanocrystals. *Nano Lett.* 8, 1374–1381 (2008).
- 35. F. Masia, W. Langbein, I. Moreels, Z. Hens, P. Borri, Exciton dephasing in lead sulfide quantum dots by *X*-point phonons. *Phys. Rev. B* **83**, 201309 (2011).
- 36. F. Gesuele, M. Y. Sfeir, W. K. Koh, C. B. Murray, T. F. Heinz, C. W. Wong, Ultrafast supercontinuum spectroscopy of carrier multiplication and biexcitonic effects in excited states of PbS quantum dots. *Nano Lett.* 12, 2658–2664 (2012).
- J. M. An, A. Franceschetti, A. Zunger, The excitonic exchange splitting and radiative lifetime in PbSe quantum dots. *Nano Lett.* 7, 2129–2135 (2007).
- 38. Z. Hu, Y. Kim, S. Krishnamurthy, I. D. Avdee, M. O. Nestoklon, A. S. Singh, A. V. Malko, S. V. Goupalov, J. A. Hollingsworth, H. Htoon, Intrinsic exciton photophysics of PbS quantum dots revealed by low-temperature single nanocrystal spectroscopy. *Nano Lett.* **19**, 8519–8525 (2019).
- 39. M. T. Trinh, M. Y. Sfeir, J. J. Choi, J. S. Owen, X. Zhu, A hot electron-hole pair breaks the symmetry of a semiconductor quantum dot. *Nano Lett.* **13**, 6091–6097 (2013).
- 40. T. J. Liptay, L. F. Marshall, P. S. Rao, R. J. Ram, M. G. Bawendi, Anomalous Stokes shift in CdSe nanocrystals. *Phys. Rev. B* 76, 155314 (2007).
- 41. A. N. Poddubny, M. O. Nestoklon, S. V. Goupalov, Anomalous suppression of valley splittings in lead salt nanocrystals without inversion center. *Phys. Rev. B* **86**, 035324 (2012).
- 42. W. Langbein, J. M. Hvam, Biexcitonic bound and continuum states of homogeneously and inhomogeneously broadened exciton resonances. *Phys. Status Solidi A* **190**, 167–174 (2002).

- T. Ghosh, J. Dehnel, M. Fabian, E. Lifshitz, R. Baer, S. Ruhman, Spin blockades to relaxation of hot multiexcitons in nanocrystals. *J. Phys. Chem. Lett.* 10, 2341–2348 (2019).
- 44. M. S. Azzaro, A. K. Le, H. Wang, S. T. Roberts, Ligand-enhanced energy transport in nanocrystal solids viewed with two-dimensional electronic spectroscopy. *J. Phys. Chem. Lett.* **10**, 5602–5608 (2019).
- 45. T. Förster, Fluoreszenzspektrum und Wasserstoffionen-konzentration. *Naturwissenschaften* **36**, 186–187 (1949).
- 46. R. A. Marcus, Relation between charge transfer absorption and fluorescence spectra and the inverted region. *J. Phys. Chem.* **93**, 3078–3086 (1989).
- 47. S. R. de Groot, P. Mazur, Non-Equilibrium Thermodynamics (Dover, 1984).
- 48. G. D. Scholes, G. R. Fleming, L. X. Chen, A. Aspuru-Guzik, A. Buchleitner, D. F. Coker, G. S. Engel, R. van Grondelle, A. Ishizaki, D. M. Jonas, J. S. Lundeen, J. K. McCusker, Shaul Mukamel, J. P. Ogilvie, A. Olaya-Castro, M. A. Ratner, F. C. Spano, K. B. Whaley, X. Zhu, Utilizing coherence to enhance function in chemical and biophysical systems. *Nature* 543, 647–656 (2017).
- 49. D. M. Jonas, Vibrational and nonadiabatic coherence in 2D electronic spectroscopy, the Jahn-Teller effect, and energy transfer. *Annu. Rev. Phys. Chem.* **69**, 327–352 (2018).
- 50. J. Cao, R. J. Cogdell, D. F. Coker, H.-G. Duan, J. Hauer, U. Kleinekathöfer, T. L. C. Jansen, T. Mančal, R. J. D. Miller, J. P. Ogilvie, V. I. Prokhorenko, T. Renger, H.-S. Tan, R. Tempelaar, M. Thorwart, E. Thyrhaug, S. Westenhoff, D. Zigmantas, Quantum biology revisited. *Sci. Adv.* 6, eaaz4888 (2020).
- D. Baranov, M. J. Lynch, A. C. Curtis, A. R. Carollo, C. R. Douglass, A. M. Mateo-Tejada, D. M. Jonas, Purification of oleylamine for materials synthesis and spectroscopic diagnostics for *trans* isomers. *Chem. Mater.* **31**, 1223–1230 (2019).
- 52. J. D. Ingle, S. R. Crouch, Spectrochemical Analysis (Prentice-Hall, 1988).

- I. Ketskeméty, J. Dombi, R. Horvai, Fluoreszenzemission, Absorption und Temperaturstrahlung von Lösungen. Ann. Phys. 463, 342–352 (1961).
- O. E. Semonin, J. C. Johnson, J. M. Luther, A. G. Midgett, A. J. Nozik, M. C. Beard, Absolute photoluminescence quantum yields of IR-26 Dye, PbS, and PbSe quantum dots. *J. Phys. Chem. Lett.* 1, 2445–2450 (2010).
- 55. G. A. Crosby, J. N. Demas, Measurement of photoluminescence quantum yields. Review. *J. Phys. Chem.* **75**, 991–1024 (1971).
- E. H. Kennard, On the interaction of radiation with matter and on fluorescent exciting power. *Phys. Rev.* 28, 672–683 (1926).
- 57. I. Moreels, K. Lambert, D. Smeets, D. De Muynck, T. Nollet, J. C. Martins, F. Vanhaecke, A. Vantomme, C. Delerue, G. Allan, Z. Hens, Size-dependent optical properties of colloidal PbS quantum dots. *ACS Nano* 3, 3023–3030 (2009).
- 58. D. Brida, S. Bonora, C. Manzoni, M. Marangoni, P. Villoresi, S. De Silvestri, G. Cerullo, Generation of 8.5-fs pulses at 1.3 μm for ultrabroadband pump-probe spectroscopy. *Opt. Express* 17, 12510– 12515 (2009).
- E. Zeek, K. Maginnis, S. Backus, U. Russek, M. Murnane, G. Mourou, H. Kapteyn, G. Vdovin, Pulse compression by use of deformable mirrors. *Opt. Lett.* 24, 493–495 (1999).
- 60. S. M. Gallagher Faeder, D. M. Jonas, Two-dimensional electronic correlation and relaxation spectra: Theory and model calculations. *J. Phys. Chem. A* **103**, 10489–10505 (1999).
- E. M. Grumstrup, S.-H. Shim, M. A. Montgomery, N. H. Damrauer, M. T. Zanni, Facile collection of two-dimensional electronic spectra using femtosecond pulse-shaping technology. *Opt. Express* 15, 16681–16689 (2007).
- 62. F. D. Fuller, J. P. Ogilvie, Experimental implementations of two-dimensional Fourier transform electronic spectroscopy. *Annu. Rev. Phys. Chem.* **66**, 667–690 (2015).

- 63. R. J. Bell, Introductory Fourier Transform Spectroscopy (Academic Press, 1972).
- J. D. Hybl, A. Albrecht Ferro, D. M. Jonas, Two-dimensional Fourier transform electronic spectroscopy. J. Chem. Phys. 115, 6606–6622 (2001).
- M. K. Yetzbacher, N. Belabas, K. A. Kitney, D. M. Jonas, Propagation, beam geometry, and detection distortions of peak shapes in two-dimensional Fourier transform spectra. *J. Chem. Phys.* 126, 044511 (2007).
- 66. A. P. Spencer, H. Li, S. T. Cundiff, D. M. Jonas, Pulse propagation effects in optical 2D Fouriertransform spectroscopy: Theory. *J. Phys. Chem. A* **119**, 3936–3960 (2015).
- 67. J. Ryu, thesis, University of Colorado, Boulder (2018).
- N. C. Anderson, M. P. Hendricks, J. J. Choi, J. S. Owen, Ligand exchange and the stoichiometry of metal chalcogenide nanocrystals: Spectroscopic observation of facile metal-carboxylate displacement and binding. *J. Am. Chem. Soc.* 135, 18536–18548 (2013).
- 69. D. Segets, J. M. Lucas, R. N. Klupp Taylor, M. Scheele, H. Zheng, A. P. Alivisatos, W. Peukert, Determination of the quantum dot band gap dependence on particle size from optical absorbance and transmission electron microscopy measurements. *ACS Nano* 6, 9021–9032 (2012).
- R. Ihly, J. Tolentino, Y. Liu, M. Gibbs, M. Law, The photothermal stability of PbS quantum dot solids. *ACS Nano* 5, 8175–8186 (2011).
- 71. M. Chemnitz, thesis, Friedrich-Schiller Universität Jena, (2019).
- 72. O. I. Micic, H. M. Cheong, H. Fu, A. Zunger, J. R. Sprague, A. Mascarenhas, A. J. Nozik, Sizedependent spectroscopy of InP quantum dots. *J. Phys. Chem. B* **101**, 4904–4912 (1997).
- T. Förster, in *Biological Physics*, E. V. Mielczarek, E. Greenbaum, R. S. Knox, Eds. (American Institute of Physics, 1993), pp. 148–160.
- T. Förster, in *Modern Quantum Chemistry*, O. Sinanoğlu, Ed. (Academic Press Inc., 1965), vol. III, pp. 93–137.

- 75. S. E. Braslavsky, E. Fron, H. B. Rodríguez, E. San Román, G. D. Scholes, G. Schweitzer, B. Valeur, J. Wirz, Pitfalls and limitations in the practical use of Förster's theory of resonance energy transfer. *Photochem. Photobiol. Sci.* 7, 1444–1448 (2008).
- R. J. Hill, T. L. Courtney, S. D. Park, D. M. Jonas, Lightweight hollow rooftop mirrors for stabilized interferometry. *Opt. Eng.* 52, 105103–105103 (2013).
- 77. D. J. Jones, E. O. Potma, J.-x. Cheng, B. Burfeindt, Y. Pang, J. Ye, X. S. Xie, Synchronization of two passively mode-locked, picosecond lasers within 20 fs for coherent anti-Stokes Raman scattering microscopy. *Rev. Sci. Instrum.* 73, 2843–2848 (2002).
- 78. R. N. Bracewell, *The Fourier Transform and its Applications* (McGraw-Hill series in Electrical and Computer Engineering Circuits and Systems, McGraw Hill, ed. 3, 2000).
- 79. S. T. Roberts, J. J. Loparo, A. Tokmakoff, Characterization of spectral diffusion from twodimensional line shapes. *J. Chem. Phys.* **125**, 084502 (2006).
- 80. P. A. Jansson, Deconvolution: With Applications in Spectroscopy (Academic Press, 1984).
- Y. Choi, S. Sim, S. C. Lim, Y. H. Lee, H. Choi, Ultrafast biexciton spectroscopy in semiconductor quantum dots: Evidence for early emergence of multiple-exciton generation. *Sci. Rep.* 3, 3206 (2013).
- P. R. Bevington, *Data Reduction and Error Analysis for the Physical Sciences* (McGraw-Hill, ed. 3, 2003).
- 83. A. Rohatgi, "WebPlotDigitizer" version 4.4 (2020); https://automeris.io/WebPlotDigitizer.
- S. Mukamel, Femtosecond optical spectroscopy: A direct look at elementary chemical events. *Annu. Rev. Phys. Chem.* 41, 647–681 (1990).