

Precise Control of Nanoscale Cu Etching via Gas-Phase Oxidation and Chemical Complexation

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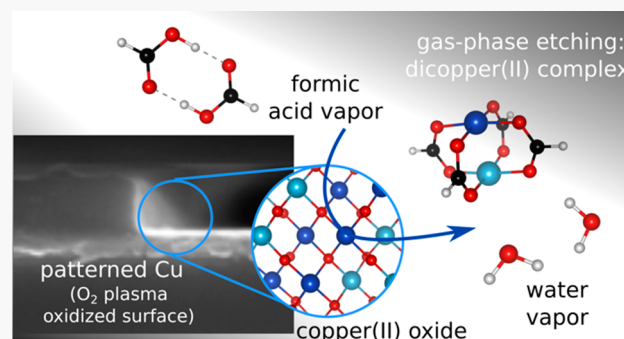
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ABSTRACT: We present a cyclic process for selective and anisotropic atomic layer etching of copper: an oxygen plasma modulates the depth and directionality of the oxidized layer, while formic acid vapor selectively removes the copper oxide scale from the metallic copper. Via density functional theory, with finite temperature and pressure free energy corrections, we evaluate the feasibility of formation of gas-phase Cu(II) and Cu(I) complexes with formate, water, formic acid, and combinations thereof as ligands. These complexes result from the neutralization reaction between copper oxide (CuO and Cu₂O) and formic acid, with and without water. We identified and evaluated the formation free energies of formate, formic acid, aquahydroxo, and aquaformate complexes of Cu(II) and Cu(I). Under relevant experimental pressures, we find the water-free dimeric tetra(μ -formate)dicopper(II) "paddlewheel" complex (Cu₂(HCOO)₄) to be the most favorable etching product, with its formation reaching equilibrium conditions from CuO. The most likely precursor for the dimer is the diformate di(formic acid)copper(II) monomer, which favorably dimerizes under the same water-lean condition at which the dimer persists. Stabilization of gas-phase Cu (oxide) derivatives thus can be achieved through complexation, enabling gas-phase etching of Cu. This work provides complementary experimental and theoretical studies that illuminate the nature of highly controlled etching with formic acid of nanoscopic CuO(s) layers covering Cu nanoarchitectures, which is relevant for the fabrication of next-generation integrated circuits.



1. INTRODUCTION

Copper (Cu) has been the standard metal for interconnects in integrated circuits (ICs) since the late 1990s due to its superior conductivity and reliability, enabling smaller and faster microprocessors.¹ The perceived challenges in finding an effective, low-temperature, gas-phase etching process to pattern Cu led to the development of the Damascene and chemical-mechanical polishing (CMP) processes in which the interlayer dielectrics are patterned lithographically but do not require etching of copper.² While these processes eliminated the need to etch copper, other challenges arose, such as high-aspect-ratio trenches filling uniformity³ and CMP-induced stress cracking.⁴ Cu was predicted to be etchable at lower temperatures by thermodynamic analysis⁵ and was achieved experimentally,^{6–8} but Damascene and CMP have remained mainstream for copper integration. In recent years, 13–15 levels of metal layers are typically required to support technology nodes beyond 10 nm.⁹ Figure 1 shows a high-performance, dense interconnect that meets computing needs and reliability requirements.¹⁰ This top-heavy structure clearly constrains the fragile low-*k* material and creates a new challenge related to removing the copper overburden, which

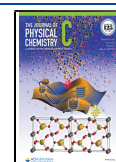
reinforces the need for a selective, low-temperature, gas-phase process to etch copper.¹¹ The most viable strategy is to develop a chemically controlled atomic layer etching (ALE) process that can pattern copper without grain-size effects and that has high selectivity toward the barrier layer.

ALE is sometimes simplistically referred to as reverse atomic layer deposition (ALD). Though that is inaccurate, intuitively, ALD studies often provide insight into what chemistries might be viable for ALE, especially in identifying potential volatile reaction products (which may be used as precursors for ALD). ALD has been instrumental in the synthesis of metals involving an oxidized metal and a reductant, "packaged" together as a gas-phase complex. At high temperatures, the metal cation is reduced by the reductant ligand, thereby depositing the metal.

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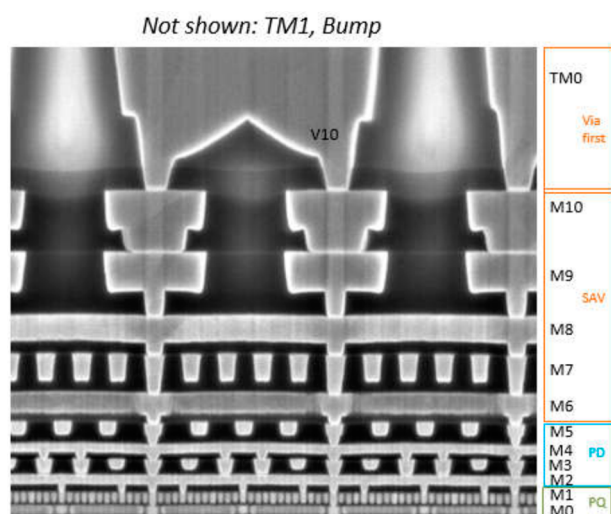


Figure 1. Illustration of the lowest 10 layers of metal (copper) of a 10 nm interconnect stack.¹⁰ M = metal, TM = thick metal, V = via, PQ = pitched quartered, PD = pitch doubled, SAV = self-aligned via. Reprinted with permission from ref 10. Copyright 2018 IEEE.

Therefore, it is conceivable to attempt the reverse process to etch metals, where the metal surface is first oxidized, and then neutralized and complexed with a gaseous organic acid etchant. From an electrochemical standpoint, Cu(II) has a standard reduction potential to Cu(0) of 0.3419 V vs the standard hydrogen electrode (SHE), making it one of the most readily reducible metal cations to its elemental form. It is even easier for Cu(I) to reduce to Cu(0) (which also easily oxidizes to Cu(II)) due to its higher standard reduction potential of 0.521 V vs the SHE.¹² Although elemental copper is easily produced from its ionized forms via electrochemistry, such reduction reactions should only happen in closely controlled environments requiring reasonable molecular stability. More specifically, many organometallic precursors have been developed for the deposition of copper thin films at temperatures ranging from 100 to 390 °C, including bis(acetylacetonato)copper(II) and its fluorinated form bis(hexafluoroacetylacetonato)copper(II), bis(*N,N'*-diisopropyl-acetamidinato)copper(II), and bis-(dimethylamino-2-propoxy)copper(II).¹³ These results suggest that copper may be etchable by selected organic chemistries, with tailored oxidation states.

Computational methods based on density functional theory (DFT), with at least gradient-corrected exchange-correlation functionals, now routinely provide atomic-scale understanding of many ground-state chemical reactions and materials properties.^{14–18} DFT-based simulations have been instrumental to innovations in the fields of photovoltaics,^{19–21} (photo)-electrocatalysis,^{22–27} energy storage devices,^{28,29} and materials (bulk and surface) processing^{30–33} to name a few. It has become an enabling approach for functional material design and chemical process development and has been used to evaluate gas-phase chemical processes relevant to the semiconductor industry, including chemical vapor deposition (CVD) and ALD of copper.^{34,35} More recently, DFT has provided understanding of surface reactions during ALE of aluminum oxide and polymers.^{36,37} While ALE of copper has been reported recently experimentally,^{38–40} quantification of the volatile reaction products has been challenging due to their low concentrations. It is therefore the goal of this work to combine experimental studies and theoretical insight to

delineate the most probable reaction mechanism leading to selective ALE of copper.

Specifically, gas-phase formic acid (HCOOH) is used here to demonstrate its capability to react preferentially with Cu oxide over Cu metal at a relatively mild temperature (80 °C). Although HCOOH has been utilized as a reducing agent for CVD and ALD of Cu,^{41–45} the mechanism that leads to etching is believed to be via acid–base reaction between HCOOH and Cu oxide, resulting in the formation of gaseous neutralization products, namely, water vapor (H₂O(g)) and Cu(II)-formate complexes. Decomposition of HCOOH on Cu and Cu oxide, which generally only occurs at temperatures greater than 180 °C, simply reduces the oxide back to its metallic form and does not lead to etching.^{46,47} Leveraging the above-mentioned etching selectivity for the oxide, we have developed a two-step cyclic process, consisting of: (1) oxidation via exposure to O₂ plasma to form a surface copper-oxide layer and (2) selective removal of the oxide layer via etching with HCOOH vapor, leaving a pristine metal Cu surface behind. This process can be repeated for continual layer-by-layer removal of Cu, where the control of etch rate and anisotropy can be demonstrated through tuning of the O₂ plasma processing conditions. Exploration via DFT calculations of various gas-phase complexes of Cu(II) with formate, formic acid, and water suggests that gaseous monomeric and dimeric aquaformate and formate Cu(II) complexes are the most likely to form, rationalizing the observed oxide etching. Although experimentally CuO is not expected to reduce to Cu₂O upon exposure to formic acid at temperatures lower than ~200 °C,⁴⁸ we also investigated computationally the formation of Cu(I) complexes as etching products. Cu(I) etching was found to be more thermodynamically difficult and therefore would be less likely to participate in etching even if present.

2. METHODS

2.1. Experiment. Two Cu blanket thin films were utilized to assess the effectiveness of the etching processes, both prepared by sputter-deposition, one with a thickness of 70 nm and one with a thickness of 25 nm. The thicker sample was used for solution-phase etching to more easily quantify the etch rate. The thinner sample was used for ALE processing. Copper oxide thin films were obtained by heating the thicker Cu stack at 150 °C for 1 h in air, which led to an observed volumetric expansion of around 60%, in line with the Pilling–Bedworth ratio for Cu. This annealing process leads to the formation of mainly CuO with some Cu(OH)₂ (CuO_x is used to collectively denote the oxide formed). Additionally, one patterned Cu sample was used to evaluate the etching anisotropy and profile control: a 55 nm Cu sample, with a Ta barrier layer with a 135 nm thick SiCN hard mask that is patterned into 6.5 and 1.5 μm lines with 4.5 and 2 μm spacings. The atomic composition and chemical states of Cu were obtained using X-ray photoelectron spectroscopy (XPS, Kratos XPS Axis Ultra DLD, with 40 eV passing energy). To quantify the etch rate, the thickness of Cu was imaged by cross-sectional scanning electron microscopy (SEM, FEI Nova 600, with 10 keV electron beam energy), before and after the etching process. The line scan route is labeled on top-down SEM images to compare the spatial distribution of elements of interest between samples before and after treatments. The elemental distribution on patterned samples were examined by energy dispersive spectroscopy (EDS, FEI NOVA 230, with 10 keV electron beam energy).

For solution-phase etching, stock solutions of HCOOH (95%, Sigma-Aldrich) were maintained at a constant temperature of 80 °C utilizing a water bath. Cu and CuO_x samples were submerged in the stock solutions for a specified period of time, rinsed in deionized H₂O and dried by nitrogen (ultrapurity (UHP), Airgas).

For gas-phase etching, a cyclic oxidation–formic acid exposure process was developed. For the oxidation half-cycle, an oxygen (UHP, Airgas) plasma was generated at 30 mTorr in an etcher (Unaxis 770) with a radio frequency (rf) source power of 500 W. The samples were placed on a carrier wafer which received an applied bias ranging from 0 to 50 W, with no additional heating, for 1–2 min. For the formic acid half-cycle, a custom-built, hot-walled, vacuum reactor homemade with a base pressure of 0.8 Torr was used. To introduce HCOOH into the chamber, high-purity nitrogen (UHP, Airgas, 100 sccm) and HCOOH (95%, Sigma-Aldrich, in a glass ampule) at a flow rate of 0.1 g/min were introduced into a temperature-controlled vaporizer (HORIBA MI-1000) at 80 °C. In the chamber, the total pressure was kept at 250 Torr. The total chamber pressure is attributed primarily to the etchant mixture (HCOOH+N₂) and therefore used as a basis for determination of the pressure of the etchant during the formic acid half-cycle. For DFT simulations (vide infra) two pressures were studied, namely, 250 and 125 Torr as upper and lower bounds of the vapor pressure of HCOOH, respectively. The samples were placed in the chamber where the chamber wall was held at 80 °C for all experiments. These two process steps (oxidation and formic acid exposure) can be repeated as a cycle, with a 5 min vacuum break between oxidation and formic acid vapor exposure.

2.2. Theory and Simulation. All of the simulations were performed within the projector augmented-wave (PAW)⁴⁹ periodic-planewave DFT formalism as implemented in the Vienna Ab-initio Simulation Package (VASP) version 5.4.4.⁵⁰ The Perdew–Burke–Ernzerhof (PBE) exchange–correlation functional was used in the evaluation of the Kohn–Sham electronic energies.⁵¹ PAW potentials were used for all elements, with frozen core orbitals of 1s for C and O, and (1s2s2p3s3p) for Cu. The simplified rotationally invariant effective Hubbard *U* correction⁵² (+*U*) was included for Cu 3d orbitals (*U* = 3.6 eV, derived from unrestricted Hartree–Fock theory on electrostatically embedded Cu₂O clusters⁵³). van der Waals interactions were simulated via the D3 method with Becke–Johnson damping (D3BJ).^{54,55} A 660 eV planewave basis kinetic-energy cutoff was imposed. A Gaussian electronic smearing (0.01 eV) was applied. The electron spin of Cu(II) was initialized with one unpaired electron, as expected for a valence d⁹ transition metal. The Cu(II) ion(s) in all species, bulk and molecular complexes, converged to an absolute projected net electronic spin of approximately 0.6–0.7 *e*, thus confirming the existence of Cu(II). Cu(I)-containing species were simulated spin-unpolarized, since Cu(I) has a closed-shell d¹⁰ electronic valence.

2.2.1. Solid-Phase Oxides. An absolute atomic force threshold of 0.002 eV/Å was imposed during lattice vector and atomic-position optimizations. To reduce numerical noise in forces and stresses, the fine-grid fast Fourier transform (FFT) was set to 4 times the coarse FFT, and the PAW core compensation charge was evaluated at 8 times the default grid. The converged lattice vectors of the various antiferromagnetic (AFM) and ferromagnetic (FM) monoclinic CuO (Figure S1), orthorhombic Cu(OH)₂ (Figure S2), and their respective

Brillouin zone (*k*-point) sampling, are summarized in Tables S1 and S2. The pristine cubic Cu₂O has no unpaired electrons; therefore, no optimization of the magnetic structure was performed. The Cu(I) oxide's optimized lattice parameters and accompanying *k*-point sampling are in Table S3. The phonon frequencies of the three solids were calculated by generating multiple finite atomic displacements (±0.01 Å from equilibrium) using phonopy (version 2.4.1)⁵⁶ and calculating atomic forces using VASP. The subsequent postprocessing to obtain the phonon densities of states (phDOS) and vibrational free energy contributions (Table S4) were performed also using phonopy. The solids' phDOS (Figure S3) confirm the stability of the optimized crystal structures found.

2.2.2. Gas-phase molecules. A periodic box size of 22 × 22 × 18 Å³ was used to isolate the gas-phase HCOOH, (HCOOH)₂, H₂O, CO₂ molecules and Cu(II) and Cu(I) complexes (vide infra), with a dipole correction applied in all directions, which include corrections for both the energy and potential.^{57,58} A Γ -point-only *k*-point sampling was employed. An absolute atomic force threshold of 0.01 eV/Å was imposed for atomic structure optimization. The vibrational frequencies were evaluated within the harmonic approximation, with the Hessians constructed via the finite atomic displacement method (±0.02 Å from equilibrium). The calculation of the free energy contributions from the translational, rotational, and vibrational degrees of freedom are described in the Supporting Information, methods section I (Tables S4 and S5). Benchmarking of the same DFT parameters used here for both gaseous HCOOH and (HCOOH)₂ are available in ref 59.

The free energy of the gaseous O₂ molecule at temperature $T(G(O_2(g), T))$ was calculated from the DFT energy of an O atom (E_O^{DFT} , spin = 2 μ_B , simulated in a 22 × 20 × 18 Å³ box), its translational free energy at $T(G_{trans}(O(g), T))$, Supporting Information, methods section I; Tables S4 and S5), and the experimental formation free energy of gaseous O atom at $T(\Delta G_f^{exp}(O(g), T))$, taken from the NIST-JANAF database:⁶⁰

$$G(O_2(g), T) = 2(E_O^{DFT} + G_{trans}(O(g), T) - \Delta G_f^{exp}(O(g), T))$$

The O₂ molecule was not explicitly modeled in DFT because PBE is known to overstabilize the molecule; it overestimates the molecule's atomization energy by as much as 1 eV/O₂.⁵¹

3. RESULTS AND DISCUSSION

3.1. Etching Copper in Organic Solutions. To determine the etch selectivity of Cu compared to CuO_x via reaction with formic acid, solution-phase etching was first performed. Figure 2 shows a plot of film thickness as a function of time for Cu and CuO (CuO_x sample thickness is 110 nm due to expansion from thermal oxidation); it is evident that the etch rate is significantly higher for the oxide as compared to the metal. Figure 2 shows the evolution of residual CuO_x as a function of etch time. Due to the inhomogeneity of the solution phase etch (Figure S4) at very short times, an accurate etch rate was difficult to determine from the slope of the curve. However, a conservative etch rate can be estimated at the longest etch time (10 s) to be ~520 nm/min. In comparison to the etch rate of Cu (1.7 nm/min), the selectivity would be >300 (calculated from the ratio of the etch rates). Although the selectivity is large for etching CuO over Cu in HCOOH solution, the use of a solution process is not desired due to the isotropic nature of the etching process that can undercut the

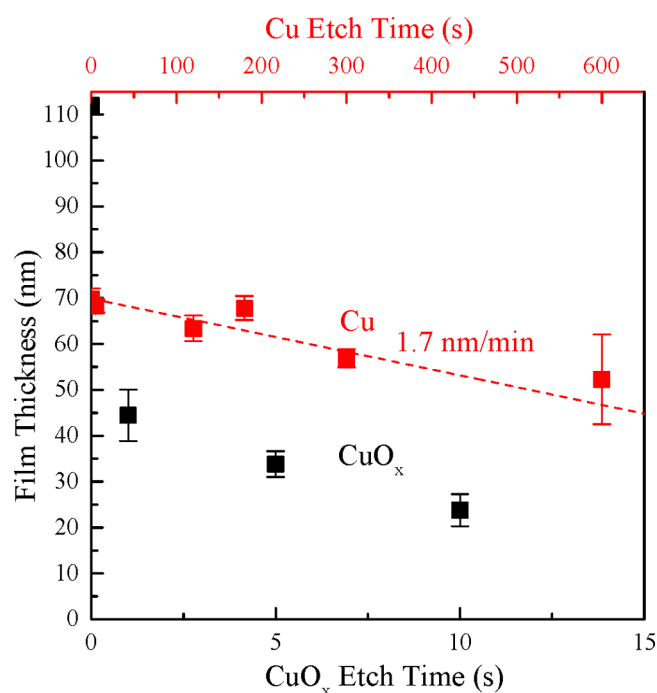


Figure 2. Remaining film thickness as a function of etching time for copper and copper oxide in formic acid solution at 80 °C. The Cu sample consisted of a 70 nm Cu layer deposited on a 15 nm TiN layer on Si. Thermal oxidation of the 70 nm Cu layer was performed via annealing under ambient conditions, where the film had expanded to ~110 nm post thermal oxidation. The selectivity for etching CuO over Cu in HCOOH solution is >300. Note the difference in red versus black time scales.

features under a mask. It is also difficult to completely remove the remaining chemical, due to the surface tension effect, from the nanoscale-patterned surface features.

3.2. Etching Copper by Gas Phase Organics. Having established the selectivity of etching copper oxide versus copper metal to be >300 in the HCOOH solution, the corresponding selectivity in HCOOH vapor was determined. Figure 3 shows high-resolution XPS spectra over the Cu 2p region for the (a) as-deposited Cu as well as (b) a sample treated with 250 Torr HCOOH (with N₂ as carrier gas) for 20 min at 80 °C. XPS was performed ex situ, but air exposure was minimized between HCOOH exposure and characterization. As can be seen in the XPS spectra, the reduction of Cu 2p_{3/2} peaks associated with Cu(OH)₂ (934.5 eV) and CuO (933.5 eV)⁶¹ as well as Cu(II) 2p_{3/2} satellite (sat.) features at 944.5 eV was consistent with the removal of Cu(II) oxide. Cu₂O and Cu could not be resolved at the resolution of the spectrometer (difference ~0.1 eV), where Cu₂O has been observed to form first upon native oxide formation.⁶² To better distinguish different Cu species, the Cu LMM Auger line also was examined; the transition from Cu⁰ to Cu²⁺ post oxidation was clearly observed (Figure S5). Imaging by SEM of both samples (not shown) confirmed that only the native oxide was removed and that no measurable etching of the metallic copper was observed.

To leverage the etch selectivity of Cu via CuO_x with HCOOH vapor, an ALE approach was taken, consisting of plasma oxidation followed by self-limiting removal of the oxide layer by HCOOH vapor. Therefore, controlled etch and subsequent Cu removal can be finely controlled by the

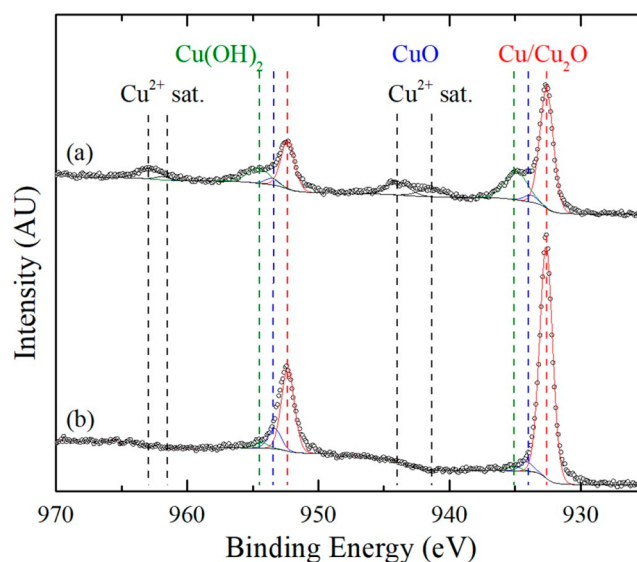


Figure 3. XPS Cu 2p spectra from a 25 nm Cu film in the (a) as-deposited state and (b) post-HCOOH-vapor-exposure (20 min, 250 Torr (HCOOH, under N₂), 80 °C). The copper feature was deconvoluted into Cu(0), Cu(I), Cu(II), and Cu(OH)₂ states, along with satellite (sat.) peaks. The copper native oxide layer was removed by the formic acid vapor but metallic copper remained.

oxidation step in which O₂ plasma as an oxidant modulates the etch anisotropy through substrate biasing. The process approaches self-limiting behavior for both half-steps when the oxidation conditions produce a diffusion-limited oxide layer. A cyclic etch process consisting of an O₂ plasma oxidation step (500 W rf power, 0 W bias, 2 min) and a formic acid vapor step (250 Torr HCOOH (with N₂) at 80 °C, 5 min) was performed for several cycles on a 25 nm Cu film. An etch rate of 2.4 nm/cycle was observed at the conditions tested (Figure 4a). Figure 4b shows the high-resolution XPS spectra of the Cu 2p region for the as-deposited film and after 4 and 8 cycles of ALE, where the Cu was nearly completely removed.

3.3. Etching of Patterned Copper Thin Film. The process described above was then applied to patterned Cu samples to assess profile control. The patterned sample was first treated with a 2 min oxidation for cleaning mask opening residue at 800 W rf and 70 W applied bias. During this pretreatment process, around 55 nm of the hard mask was removed, but the Cu layer was not etched. Figure 4 shows the comparison between the as-patterned Cu thin film before and after 5 cycles consisting of an oxidation step (500 W rf power and 0 W applied bias for 2 min) and a HCOOH (with N₂) exposure (250 Torr, 80 °C for 5 min). The cross-sectional SEM images (Figure 5a,b) show that Cu in the exposed region was completely removed, with limited change in the hard mask thickness during the cyclic process utilizing oxygen plasma at lower power and bias. Top-down SEM images (Figure 5c,d) were taken of regions in close proximity before and after the process to compare the profile; the brighter regions in the pretreatment sample image correspond to the exposed Cu while the darker regions are those blocked by the SiCN hard mask. EDS line scans (Figure 5e,f), along highlighted arrows in Figure 5c,d, chemically confirmed the removal of Cu from the sample surface. The Cu signal intensity, denoted by the black curve, decreased drastically from the masked to the exposed region on the treated sample, as compared to the uniform intensity observed for the as-deposited sample. Partial removal

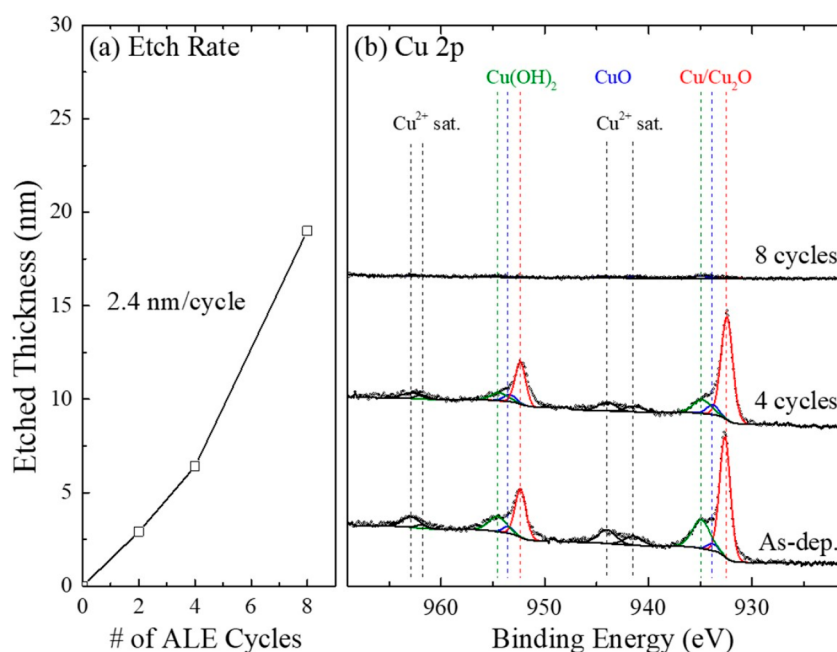


Figure 4. (a) Thickness change of 25 nm Cu by cycles of O₂ plasma and formic acid vapor, as determined via cross-section SEM with 2 min oxidation (500 W, 0 W bias, 35 mTorr) and 5 min HCOOH vapor exposure (250 Torr (formic acid with N₂), 80 °C). (b) Corresponding high-resolution XPS spectra of the Cu 2p spectra, confirming the removal of copper.

of the SiCN hard mask was noted from the SEM images before and after etching, which is not a concern for verifying the validity of the etching chemistry since the line width remains constant and the underlying Cu was protected from oxidation and etching.

3.4. DFT Simulations. The relatively large volume of the chamber compared to the miniscule amounts of etched material made direct characterization of the etching products difficult. HCOOH vapor clearly directly interacts with the oxide layer, CuO, and Cu(OH)₂; however, the reaction product and its fate in the chamber is unknown. As in the solution-phase etching, the CuO (and Cu(OH)₂) is expected to react with HCOOH via an acid–base (neutralization) reaction, followed by complexation with either water and hydroxide, unreacted HCOOH and its conjugate base formate (HCOO[−]), or some combination. Thermodynamics indicate that HCOOH also may act as a reducing agent, enabling the formation of Cu₂O from CuO. The Cu LMM Auger line XPS spectra in Figure S5 show no Cu₂O after plasma O₂ oxidation, but it suggests its emergence after etching (it is also possible that the vacuum break prior to the XPS measurement post etching may have introduced the Cu(I) species along with the hydration of residual CuO to Cu(OH)₂). Below, we present first the complexation and etching of Cu(II) from CuO and Cu(OH)₂. This is followed by an exploration of the possible participation of Cu(I) species, from Cu₂O, in the formation of volatile Cu species. Lastly, we assess the thermodynamics of the formation of gaseous neutral Cu(II) and Cu(I) molecular complexes under likely experimental conditions.

3.4.1. Etching through Neutralization and Complexation of Cu(II)OH_x. To explore the gas-phase complexes of Cu(II) with water/hydroxide, formic acid, and HCOO[−], various Cu(II) complexes were investigated by means of DFT approximations. Figure 6 summarizes the optimized structures of the various complexes that may form, which include both Cu(II) monomer and dimer complexes. Note that only neutral

complexes were studied, because unlike in aqueous phase generation of gaseous charged species off of the CuO/Cu(OH)₂ surface is not stabilized by a polarizable medium, e.g., bulk water, and therefore will be extremely unfavorable. Starting from the monoclinic AFM CuO, Figure 6 shows in sequence the formation of complexes, growing from two coordinating ligands to five (blue paths, a–d), and from monomer to dimer (violet paths, f–h). Also in the figure is the reaction connecting CuO and Cu(OH)₂ (orange path, j). The AFM structures of CuO and Cu(OH)₂ are shown through two different blue-colored Cu ions, one for excess electron spin-up and another for excess electron spin-down. For CuO, FM interactions persist along the [010] and [101] directions, with alternating AFM and FM interactions along the [101], as demonstrated in the literature.^{63,64} For Cu(OH)₂, FM interactions exist along [001] and [100], with alternating AFM and FM interactions along [010].

Although the complexation reactions are presented in sequence in Figure 6, they need not proceed in this manner. In this study, only the thermodynamics was assessed for the formation of these complexes, with CuO/Cu(OH)₂, water vapor, and formic acid as reactants (as a dimer (HCOOH)₂, which is the majority form of HCOOH in the gas phase,⁶⁵ Figure S6). Although kinetics may play a role in the feasibility of forming these chemical species, thermodynamics of course is needed to set the stage in determining possible chemical pathways. Table 1 lists the standard reaction free energies at three different temperatures, namely, room, experimental (*T* = 353 K (80 °C)), and “extreme” (*T* = 403 K; 29° above the boiling point of HCOOH), along with the chemical equations representing the reactions in Figure 6. Three temperatures were given to illuminate the dependence of the spontaneity of these reactions with temperature. From these standard reaction free energies, the nonstandard reaction free energies under a (likely) range of experimental reactant and product partial pressures (*vide infra*) can be extrapolated.

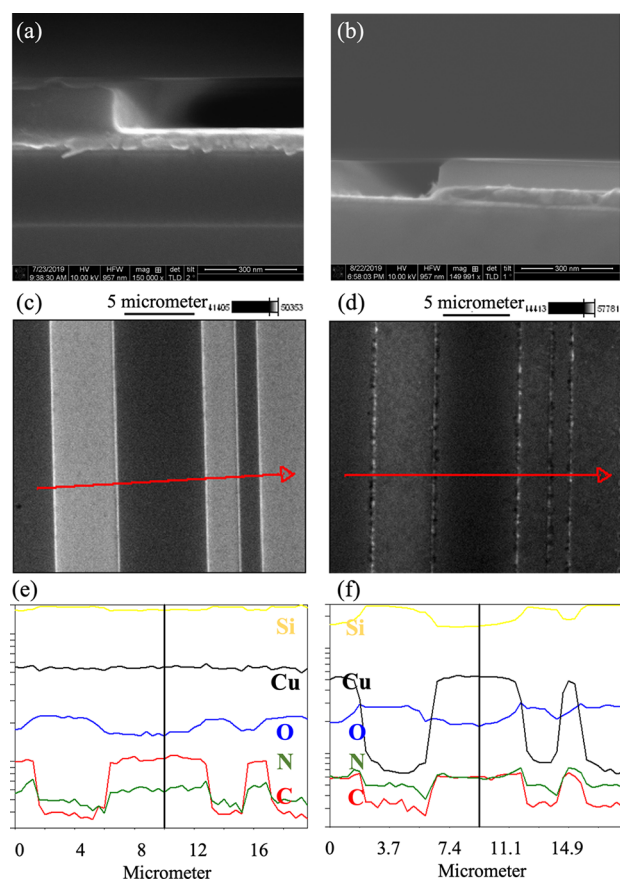


Figure 5. Characterization of the cyclic etch process (described in the main text) on patterned Cu. SEM cross-sectional image of (a) as-deposited patterned Cu and (b) processed patterned Cu. Top-down SEM of (c) as-deposited patterned Cu and (d) processed patterned Cu, with EDS line scan routes indicated by the arrows. EDS line scans along the arrows of the (e) as-deposited patterned Cu and (f) processed patterned Cu.

Reaction a in Figure 6 describes the formation of the diformatocopper(II) complex ($\text{Cu}(\kappa^2\text{-HCOO})_2$) from $(\text{HCOOH})_2$ reacting with a unit of CuO, releasing water. HCOO^- is a bidentate ligand, κ^2 , which means it has two noncontiguous coordinating atoms: the two O atoms. The two protons from $(\text{HCOOH})_2$ form a water molecule with an O in the CuO lattice (acid–base or neutralization reaction), while the resulting two HCOO^- ions coordinate with a Cu(II) ion (complexation). This reaction is highly unfavorable with a standard free energy of reaction of $\Delta G^\circ \sim 1$ eV (Table 1). $\text{Cu}(\kappa^2\text{-HCOO})_2$ exhibits a square planar structure, as in the bulk CuO (Figure 6). However, the molecule may be in a nonideal, constrained configuration. Cu(II) is a d^9 transition metal cation (one unpaired electron) that forms tetrahedral, square planar, or some intermediate coordination, the preference of which depends on the ligand and even the counterion.⁶⁷

Cu(II) can only accommodate two formate ions to maintain a neutral complex, thus to build larger neutral mononuclear complexes, only a neutral ligand, such as H_2O , can be added. Reactions b and c progressively relieve the strain in the complex by adding H_2O as ligand, converting an HCOO^- from being a bidentate (κ^2) to a monodentate (κ^1) ligand for every H_2O added: $\text{Cu}(\text{H}_2\text{O})(\kappa^1\text{-HCOO})(\kappa^2\text{-HCOO})$ or simply $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})_2$, and $\text{Cu}(\text{H}_2\text{O})_2(\kappa^1\text{-HCOO})_2$ or

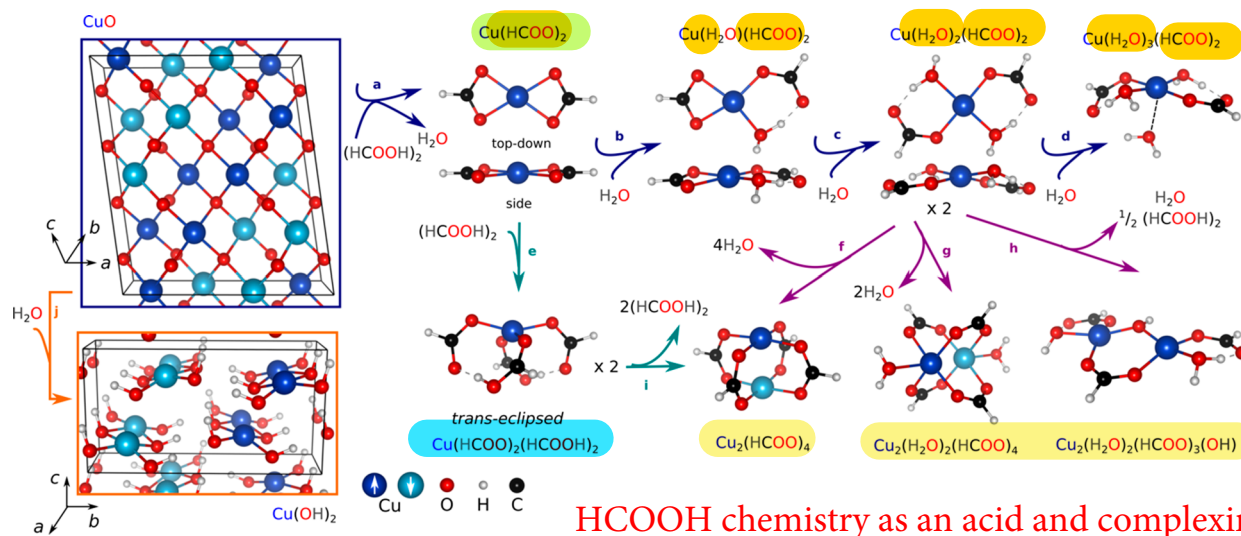
simply $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$. Furthermore, the uncoordinated O atom of the $\kappa^1\text{-HCOO}^-$ ligand forms a hydrogen bond with the coordinated H_2O . The addition of two H_2O molecules is favorable under standard conditions (Table 1), despite it being entropically unfavorable (note that the reaction becomes less spontaneous as the temperature increases). This attests to the structural (and perhaps to some degree electrostatic) nature of the significant stabilizing effect of H_2O . Indeed, in reaction d, where a third H_2O is added simply as an axial ligand, the process becomes unfavorable. This is because the third H_2O , in contrast to the first two H_2O , does not reduce the strain in the coordination of the equatorial ligands and because Cu(II) favorably accommodates only up to four ligands (the antibonding d_{z^2} -derived orbital of a square-planar-coordinated Cu(II) is fully occupied).

Higher-energy alternative structures for $\text{Cu}(\text{HCOO})_2$, $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})_2$, and $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$ complexes are shown in Figure S7. In the alternative $\text{Cu}(\text{HCOO})_2$, the two HCOO^- ions are bound via a κ^1 motif, leaving Cu(II) with only two bonds. The higher-energy $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})_2$ and $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$ in Figure S7, however, maintain the κ^2 bonding motif for HCOO^- ions, with H_2O as axial ligand(s) of Cu(II). These structures thus retain the strained square planar geometry exhibited by $\text{Cu}(\kappa^2\text{-HCOO})_2$, while also expanding the coordination number of Cu(II) beyond four, which is unfavorable.

In the absence of water, formic acid can take the place of water as a neutral ligand, with the carbonyl oxygen as the coordinating atom. Reaction e (first blue-green path) shows the formation of *trans*- $\text{Cu}(\kappa^1\text{-HCOO})_2(\text{HCOOH})_2$ in an eclipsed configuration, or simply $\text{Cu}(\text{HCOO})_2(\text{HCOOH})_2$, from $\text{Cu}(\text{HCOO})_2$. The *trans* prefix refers to the relative position of the two HCOOH , where they are opposite to each other, where the same is true for the two HCOO^- . In the complex, HCOO^- is hydrogen bonded to a neighbor HCOOH . Since the $\text{HCOOH}\cdots\text{OOCH}$ hydrogen-bonded pairs are on the same side of the (distorted) CuO_4 plane, we refer to this configuration as *eclipsed*. Two of its higher-energy isomers, the *trans-staggered* conformer (the $\text{HCOOH}\cdots\text{OOCH}$ hydrogen-bonded pairs are on the opposite sides of the CuO_4 plane) and a *cis* isomer, are presented in Figure S7. Under standard conditions, the addition of two neutral HCOOH onto $\text{Cu}(\text{HCOO})_2$, much like the addition of two water molecules, is favorable (Table 1, reaction e). The standard free energy of this addition reaction is -0.33 eV at 298 K, just 0.14 eV less negative than that of the addition of two waters, which is -0.47 eV at 298 K (Table 1, b + c).

The possibility of forming gaseous aquahydroxo complexes from $\text{Cu}(\text{OH})_2$ (Figure S8) was examined, to rule out the need for HCOO^- to directly participate in the complexation of Cu(II). Indeed, the formation of the aquahydroxo complexes is much more difficult, with standard free energies of formation relative to $\text{Cu}(\text{OH})_2$ and $\text{H}_2\text{O}(\text{g}) > 1.6$ eV/complex at 298 K, which highlights the critical role of HCOO^- in the removal of Cu(II). This finding is consistent with the very low solubility of $\text{Cu}(\text{OH})_2$ in pure water ($K_{\text{sp}} = 2.2 \times 10^{-20}$).⁶⁷

Dimerization of the complex can satisfy the coordination sphere of Cu(II), while using less molecules as ligands and also relieving bonding strain. This would enable the reduction of the entropic penalty of producing the complex by using less gaseous reactants per Cu(II) ion etched. The bidenticity of the HCOO^- ion make dimerization possible by acting as a bridging ligand (μ). The purple paths (f–h) in Figure 6



HCOOH chemistry as an acid and complexing agent

Figure 6. DFT-predicted candidate gas-phase products from etching crystalline CuO and Cu(OH)_2 in the presence of formic acid and water vapor. Reactions are indexed from a to i. The product molecules are formate (product of reaction a) and aquaformate (b–d), and diformatodi(formic acid) (e) complexes of Cu(II) . Dinuclear formate (f) and aquaformate (g and h) complexes are products of the dimerization of $\text{Cu(H}_2\text{O)}_2(\text{HCOO})_2$ releasing water (f and g) or both water and formic acid (h). An alternative path for dimerization is through the diformatodi(formic acid) complex (i). Hydration of CuO to Cu(OH)_2 (j). Two different colors of Cu denote distinct electron spins and illustrate the AFM ordering in both CuO and Cu(OH)_2 . The dimeric Cu complexes $\text{Cu}_2(\text{HCOO})_4$ and $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_4$ prefer AFM interactions, while $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH})$ prefers a FM ordering (Table S6).

describe formation of the candidate dimers from the most stable (under standard pressure and temperature) monomer: $\text{Cu(H}_2\text{O)}_2(\kappa^1\text{-HCOO})_2$. Reaction f is the formation of the paddlewheel complex $\text{Cu}_2(\mu\text{-HCOO})_4$ or tetra(μ -formate)-dicopper(II), which sheds all four H_2O molecules from the two $\text{Cu(H}_2\text{O)}_2(\kappa^1\text{-HCOO})_2$ molecules. The four HCOO^- bridge the two Cu(II) ions with the two Cu(O)_4 planes on top of each other, forming a paddlewheel-like structure (Figure 6), hence the name. Reaction g is a similar reaction, but expels one less H_2O molecule per Cu(II) , forming a fully hydrated paddlewheel complex: $\text{Cu}_2(\text{H}_2\text{O})_2(\mu\text{-HCOO})_4$, with two H_2O as axial ligands (Figure 6). The formations of these complexes are mildly favorable (-0.11 and -0.10 eV/Cu) relative to $\text{Cu(H}_2\text{O)}_2(\kappa^1\text{-HCOO})_2$ at 353 K, which become more favorable at higher temperatures, e.g., -0.21 and -0.14 eV/Cu at 403 K, because these reactions result in a net positive formation of gaseous molecules (Table 1, reactions f and g). An alternative monomer that may serve as a precursor to $\text{Cu}_2(\mu\text{-HCOO})_4$ in the absence of water is $\text{Cu(HCOO)}_2(\text{HCOOH})_2$. Dimerization is achieved by shedding two $(\text{HCOOH})_2$ molecules per $\text{Cu}_2(\mu\text{-HCOO})_4$ (Figure 6, reaction i, second blue-green path). Similar to the aqua analogue, this dimerization reaction is exoergic (-0.15 eV/Cu at 353 K). Paddlewheel transition-metal complexes are well-documented in the literature,⁶⁸ with Cu(II) -based complexes containing large organic carboxylates achieving prominence because of its central chemical motif in many metal organic frameworks (MOFs).^{69,70} Earlier documentation of Cu(II) paddlewheel complexes involved small organic carboxylates, such as crystalline hydrated tetraacetatodicopper(II), $\text{Cu}_2(\text{H}_2\text{O})_2(\mu\text{-CH}_3\text{COO})_4$,⁷¹ and anhydrous tetrapropionatodicopper(II), $\text{Cu}_2(\mu\text{-C}_2\text{H}_5\text{COO})_4$, dissolved in chloroform.⁷² Recently, infrared multiphoton dissociation has been shown to decompose anionic copper(II) formate clusters of the formula $\text{Cu(II)}_n(\text{HCOO})_{2n+1}^-$ ($2 < n \leq 8$, generated via electrospray ionization or ESI), proceeding via favorable sequential loss (evaporation) of neutral dimeric $\text{Cu}_2(\text{HCOO})_4$ until $n = 3$, at

which point Cu(HCOO)_2 is released.⁷³ The aforementioned observation was supported by the authors' DFT-B3LYP/def2TZVP simulations.⁷³

For reaction h, $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH})$ or $\text{Cu}_2(\text{H}_2\text{O})_2(\mu\text{-HCOO})(\kappa^1\text{-HCOO})_2(\mu\text{-OH})$ forms where both an HCOO^- and an OH^- act as bridging ligands. It consists of two complexes side-by-side, instead of being on top of each other as in the paddlewheel complexes. This reaction has similar thermodynamics as f and g but exhibits weaker dependence on temperature than f and g, because it results in fewer gaseous products compared to the other two reactions.

3.4.2. Etching after Pre-Reduction of Cu(II)O into $\text{Cu(I)}_2\text{O}$. So far, we viewed HCOOH chemistry as an acid and complexing agent. In this section we explore the possibility of HCOOH acting also as a reductant. Reduction of CuO to Cu_2O with HCOOH as a reducing agent is thermodynamically feasible. In this reaction, HCOOH oxidizes into CO_2 , while two Cu(II) ions reduce to Cu(I) as the two H atoms from the decomposition of HCOOH remove an O atom from the lattice to form H_2O . The empirical standard reaction free energy at 298 K for this reaction is -0.81 eV/ CuO , while we predict it to be -0.71 eV/ CuO , an approximately 0.1 eV error (Table 1). At the experimental etching temperature (353 K) and $(\text{HCOOH})_2$ pressure (approximately within 125–250 Torr, see below for the rationale of the estimates), the reaction is predicted to remain spontaneous (Figure 7a, blue lines). We set the CO_2 and H_2O partial pressures at 0.08 Torr or 1/10 of the vacuum base pressure (0.8 Torr, this pressure was imposed prior to the introduction of $\text{HCOOH}+\text{N}_2$ mixture into the reaction chamber, section 2.2). In Figure 7a, the experimental and theoretical data at 298 K are also plotted to provide an estimate to the error of the prediction at 353 K. A competing reaction that will oxidize Cu_2O back to CuO is via reaction with molecular oxygen. The standard oxidation free energy at 298 K has been also measured to be very negative: -0.56 eV/ CuO . PBE+U+D3BJ overestimates the exoergicity of this

Table 1. Standard Reaction Free Energies (eV/Cu at Standard 750 Torr Pressure) for the Formation of Cu(II) and Cu(I) Formate, Formic Acid, and Aquaformate Complexes and Bulk Phase Reactions Involving Cu(OH)₂, CuO, and Cu₂O as Depicted in Figures 6 and 7^a

reaction index	reaction equation	298 K	353 K	403 K
a	$\text{CuO} + (\text{HCOOH})_2 \rightarrow \text{Cu}(\text{HCOO})_2 + \text{H}_2\text{O}$	0.95	0.85	0.77
b	$\text{Cu}(\text{HCOO})_2 + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})(\text{HCOO})_2$	-0.27	-0.19	-0.12
c	$\text{Cu}(\text{H}_2\text{O})(\text{HCOO})_2 + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$	-0.20	-0.11	-0.03
d	$\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2 + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})_3(\text{HCOO})_2$	0.13	0.19	0.25
e	$\text{Cu}(\text{HCOO})_2 + (\text{HCOOH})_2 \rightarrow \text{Cu}(\text{HCOOH})_2(\text{HCOO})_2$	-0.33	-0.26	-0.20
f	$\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_4 + 2\text{H}_2\text{O}$	0.00	-0.11	-0.21
g	$\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_4 + \text{H}_2\text{O}$	-0.06	-0.10	-0.14
h	$\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH}) + \frac{1}{2}\text{H}_2\text{O} + \frac{1}{4}(\text{HCOOH})_2$	-0.04	-0.06	-0.08
i	$\text{Cu}(\text{HCOOH})_2(\text{HCOO})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_4 + (\text{HCOOH})_2$	-0.14	-0.15	-0.16
j	$\text{CuO} + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{OH})_2$	0.08 ^b	0.17	0.25
M1 = a + b + c	$\text{CuO} + (\text{HCOOH})_2 + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$	0.48	0.55	0.62
M2 = a + e	$\text{CuO} + 2(\text{HCOOH})_2 \rightarrow \text{Cu}(\text{HCOOH})_2(\text{HCOO})_2 + \text{H}_2\text{O}$	0.62	0.60	0.57
D1 = M1 + f	$\text{CuO} + (\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_4 + \text{H}_2\text{O}$	0.48	0.44	0.41
D2 = M1 + g	$\text{CuO} + (\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_4$	0.41	0.45	0.48
D3 = M1 + h	$\text{CuO} + \frac{3}{4}(\text{HCOOH})_2 + \frac{1}{2}\text{H}_2\text{O} \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH})$	0.44	0.49	0.54
	$\text{CuO} + \frac{1}{4}(\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{2}\text{CO}_2 + \frac{1}{2}\text{H}_2\text{O}$	-0.71 ^c	-0.75	-0.78
	$\frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{4}\text{O}_2 \rightarrow \text{CuO}$	-0.75 ^c	-0.75 ^d	-0.76 ^d
a'	$\frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{2}(\text{HCOOH})_2 + \frac{1}{2}\text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})(\text{HCOO})$	1.34	1.32	1.29
b'	$\text{Cu}(\text{H}_2\text{O})(\text{HCOO}) + \text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})$	-0.50	-0.41	-0.33
c'	$\frac{1}{2}\text{Cu}_2\text{O} + (\text{HCOOH})_2 \rightarrow \text{Cu}(\text{HCOO})(\text{HCOOH}) + \frac{1}{2}\text{H}_2\text{O}$	0.84	0.79	0.75
d'	$\text{Cu}(\text{H}_2\text{O})(\text{HCOO}) \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_2 + \text{H}_2\text{O}$	-0.84	-0.85	-0.86
e'	$\frac{1}{2}\text{Cu}_2(\text{HCOO})_2 + \text{H}_2\text{O} \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_2$	0.10	0.16	0.22
f'	$\text{Cu}(\text{HCOO})(\text{HCOOH}) \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_2 + \frac{1}{2}(\text{HCOOH})_2$	-0.33	-0.33	-0.33
g'	$\frac{1}{2}\text{Cu}_2(\text{HCOO})_2 + \frac{1}{2}(\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2$	0.03	0.06	0.09
M1' = a' + b'	$\frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{2}(\text{HCOOH})_2 + \frac{3}{2}\text{H}_2\text{O} \rightarrow \text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})$	0.84	0.90	0.96
M2' = c'	$\frac{1}{2}\text{Cu}_2\text{O} + (\text{HCOOH})_2 \rightarrow \text{Cu}(\text{HCOO})(\text{HCOOH}) + \frac{1}{2}\text{H}_2\text{O}$	0.84	0.79	0.75
D1' = a' + d'	$\frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{2}(\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_2 + \frac{1}{2}\text{H}_2\text{O}$	0.51	0.46	0.43
D2' = D1' + e'	$\frac{1}{2}\text{Cu}_2\text{O} + \frac{1}{2}(\text{HCOOH})_2 + \frac{1}{2}\text{H}_2\text{O} \rightarrow \frac{1}{2}\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_2$	0.61	0.63	0.65
D3' = D1' + g'	$\frac{1}{2}\text{Cu}_2\text{O} + (\text{HCOOH})_2 \rightarrow \frac{1}{2}\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2 + \frac{1}{2}\text{H}_2\text{O}$	0.54	0.53	0.52

Table 1. continued

^aReactions a–h and a'–g' are combined to determine the reaction free energies in forming the most stable complexes with CuO, (HCOOH)₂, and H₂O as reactants. The relevant energies at experimental temperature of 353 K (80 °C) are bolded. ^bExperimental value is –0.16 eV calculated using the standard free energies of formation at 298.15 K of CuO, Cu(OH)₂, and H₂O(g) from ref 60, which suggests that either the CuO is overstabilized or Cu(OH)₂ is destabilized in PBE+U+D3BJ. ^cExperimental values are –0.81 eV/Cu for CuO reduction with (HCOOH)₂ and –0.56 eV/Cu for Cu₂O oxidation with O₂. They were calculated using the standard free energies of formation at 298.15 K of CuO, Cu₂O, CO₂(g), H₂O(g), and O₂(g) from ref 60, and (HCOOH)₂ from ref 66. ^dThe free energy of formation of O(g) at 350 and 400 K were alternatively used in the calculation of the free energy of O₂(g) at 353 and 403 K (see section 2.2 for the equation), which were taken from the NIST-JANAF database.⁶⁰

HCOOH chemistry as a reductant

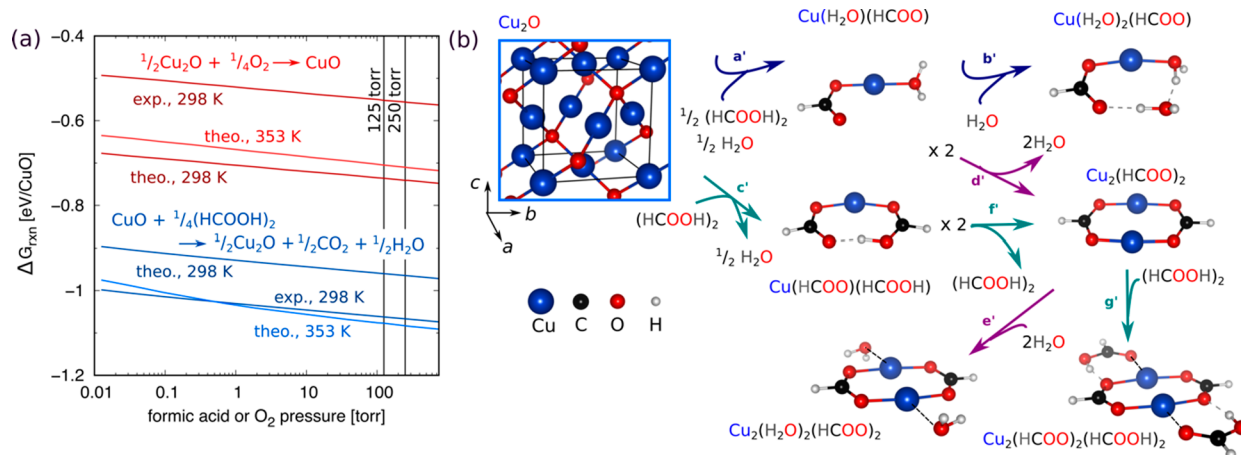


Figure 7. (a) Free energy of CuO reduction to Cu₂O, with (HCOOH)₂ as the reductant, and Cu₂O oxidation to CuO, with O₂ as the oxidant, as functions of the partial pressures of the reductant and oxidant molecules. Upper and lower bound total pressures of all HCOOH species ((HCOOH)_n): 250 and 125 Torr are marked, representing 100 and 50% (HCOOH)_n in the (HCOOH)_n+N₂ gas stream at the total exposure pressure of 250 Torr, respectively. CO₂ and H₂O partial pressures are set to 0.08 Torr (1/10 of the vacuum base pressure). (b) DFT-predicted candidate gas-phase products from etching crystalline Cu₂O in the presence of formic acid and water vapor. Reactions are indexed from a' to g'. The product molecules are aquaformate (a' and b') and formate(formic acid) (c') complexes of Cu(I). Dinuclear diformate (d') and diaquadiformate (e') complexes are products, for example, of the dimerization of Cu(H₂O)(HCOO). An alternative path for dimerization, relevant at low H₂O and high HCOOH conditions, is through Cu(HCOO)(HCOOH) (f'), which may be followed by coordination of two more HCOOH molecules forming Cu₂(HCOO)₂(HCOOH)₂ (g').

reaction to –0.75 eV/CuO, an approximately –0.2 eV error. At very low O₂ partial pressure, e.g., 0.01 Torr, and at 353 K, this reaction remains very thermodynamically favorable even when accounting for the possible error in the prediction of the oxidation free energy (Figure 7a, red lines). Cu₂O thus will dominate only when the chamber is completely depleted of O₂. The relative rates of the reductive and oxidative reactions will dictate the relative amounts of Cu(II) and Cu(I) exposed for etching. From previous thermogravimetry coupled with quadrupole mass spectroscopy experiments, reduction of copper(II) oxide nanoparticles to copper(I) oxide with formic acid was found to occur between 220–250 °C (498–523 K), which is then followed by reduction to metallic Cu at $T > 350$ °C (623 K).⁴⁸ Temperature-programmed desorption (TPD) studies also found that the CO₂ desorption peak on Cu₂O film pre-exposed to HCOOH vapor at 300 K appears at around 545 K.⁷⁴ TPD experiments on copper surfaces, from metallic Cu to CuO, showed positive correlation between the HCOOH decomposition TPD peak temperature and degree of oxidation of the surface.⁷⁴ Lower temperature reduction of CuO and Cu₂O with HCOOH only occurs in the presence of a catalyst, e.g., Pt, which can facilitate the formation of H radicals from HCOOH at a lower temperature.^{48,75} Reduction of CuO (and Cu₂O) with HCOOH thus is kinetically suppressed.

A possible combination of Cu(II) and Cu(I) is suggested in the Cu LMM Auger line XPS spectra post-etching (Figure S5). Below, we computationally explore the extent of participation

of Cu(I) in the etching process and show that Cu(I) is harder to remove than Cu(II). At the O₂ plasma step, both the residual Cu(I) and re-exposed Cu(0) will be oxidized to Cu(II), which will facilitate the removal of Cu.

We performed a similar analysis on the possible etching products of Cu₂O (Figure 7b) with the formate ion and neutral molecules, namely, water and formic acid (Figure 7b). Cu(I) is a closed-shell d¹⁰ transition-metal ion and forms a linear coordination with two lattice O in the cubic Cu₂O crystal.⁷⁶ Much like in the solid, Cu(I) complexes were also found to prefer 2-fold coordination, e.g., aquaformate Cu(I) complex, Cu(H₂O)(κ¹-HCOO) or simply Cu(H₂O)(HCOO) (Figure 7b, reaction a'). The linear coordination in Cu(H₂O)(HCOO) is found to be favored over a bent structure, although only by 0.08 eV at 298 K (Figure S9 and Table S5). Cu(H₂O)(HCOO) was found to be unable to accommodate a second H₂O molecule as a third ligand and simply binds the second H₂O molecule via hydrogen bonding with the coordinated H₂O and HCOO[–] (reaction b'). The intra-molecular hydrogen bonding in Cu(H₂O)₂(HCOO) renders the second H₂O to be bound favorably (Table 1, reaction b'). Another monomer, the formate(formic acid) Cu(I) complex, Cu(κ¹-HCOO)(κ¹-HCOOH) or Cu(HCOO)(HCOOH), exhibits interligand hydrogen bonding (reaction c'), which slightly bends the molecule. When this interligand hydrogen bonding breaks, as in a staggered ligand conformation, which

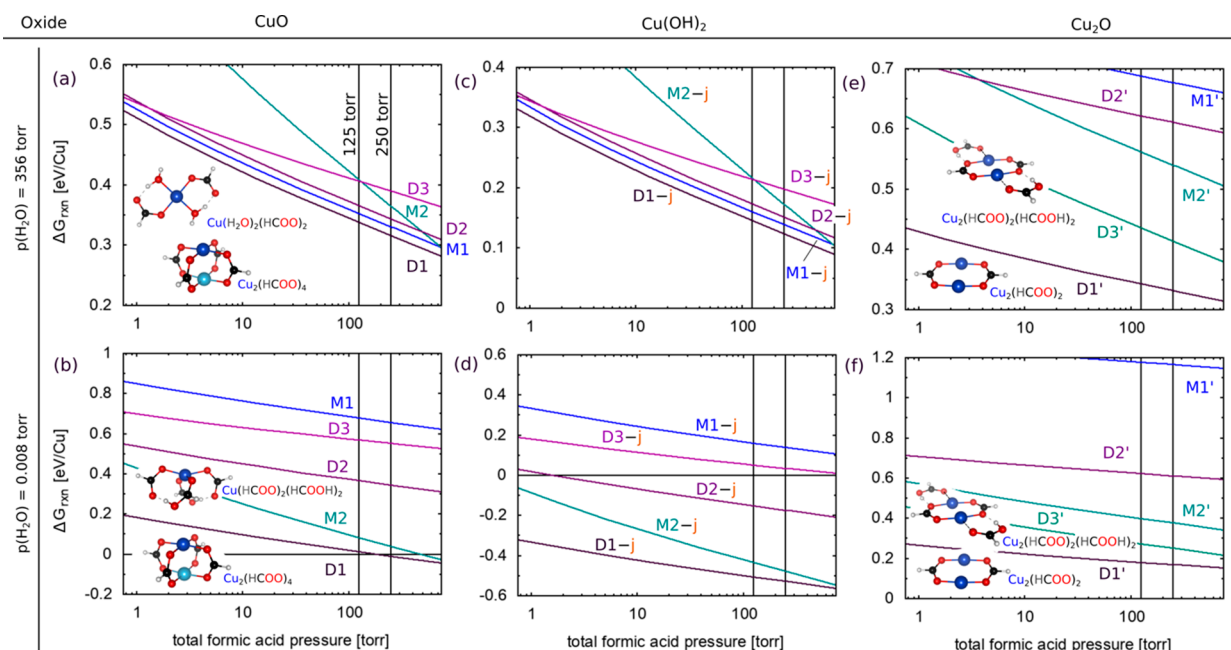


Figure 8. (a,b) and (c,d) Reaction free energy (eV/Cu) to form the most stable Cu(II) monomers: $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$ and $\text{Cu}(\text{HCOO})_2(\text{HCOOH})_2$ (reactions M1 and M2), and dimers $\text{Cu}_2(\text{HCOO})_4$, $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_4$, and $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH})$ (reactions Dm , $m = 1-3$), from CuO and $\text{Cu}(\text{OH})_2$, respectively, as a function of the total pressure of formic acid species (logarithmic scale) at 353 K (80 °C), at two different ambient water vapor pressures (356 and 8.0×10^{-3} Torr). For (c,d), j is the free energy to form $\text{Cu}(\text{OH})_2$ from CuO (Table 1). (e,f) A similar plot for the Cu(I) complexes. The most stable molecules are, namely, monomers $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})$ and $\text{Cu}(\text{HCOO})(\text{HCOOH})$ (reactions $\text{M1}'$ and $\text{M2}'$) and dimers $\text{Cu}_2(\text{HCOO})_2$, $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_2$, and $\text{Cu}(\text{HCOO})_2(\text{HCOOH})_2$ (reactions Dm' , $m = 1-3$). In all plots, a low Cu complex pressure of 8.0×10^{-2} Torr was assumed (1/10 of the vacuum base pressure). Insets show the two lowest-energy structures for each water vapor pressure condition. Upper and lower bound total pressures of all HCOOH species ($(\text{HCOOH})_n$): 250 and 125 Torr are marked, representing 100 and 50% $(\text{HCOOH})_n$ in the $(\text{HCOOH})_n + \text{N}_2$ gas stream at the total exposure pressure of 250 Torr, respectively. The plots were constructed from the standard reaction free energies in Table 1. See also Table 1 for the definition of the reaction indices.

recovers a perfectly linear configuration (Figure S9 and Table S5), its energy increases by 0.60 eV at 298 K.

Finally, we also explore the dimerization of Cu(I). Much like in Cu(II), the dimeric Cu(I) formate complex, $\text{Cu}_2(\mu\text{-HCOO})_2$, may form by dehydration of the monomeric aquaformate complexes ($\text{Cu}(\text{H}_2\text{O})(\text{HCOO})$ or $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})$). Reaction d' in Figure 7b shows the dimerization from $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})$ (-0.85 eV/Cu at 353 K, Table 1). A dimer can also form from $\text{Cu}(\text{H}_2\text{O})(\text{HCOO})$ without losing water to form instead $\text{Cu}_2(\text{H}_2\text{O})_2(\mu\text{-HCOO})_2$, corresponding to the sum of reactions d' and e' in Figure 7b (-0.68 eV/Cu at 353 K, Table 1). Under water-lean conditions, the formation of $\text{Cu}_2(\mu\text{-HCOO})_2$ may be better facilitated by the dimerization of $\text{Cu}(\text{HCOO})(\text{HCOOH})$ represented by reaction f' , Figure 7b (-0.33 eV/Cu at 353 K, Table 1). At HCOOH-rich conditions, additional HCOOH may coordinate with $\text{Cu}_2(\mu\text{-HCOO})_2$ (reaction g' , Figure 7b) to form $\text{Cu}_2(\mu\text{-HCOO})_2(\kappa^1\text{-HCOOH})_2$ or simply $\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2$. Unlike in Cu(II), $\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2$ is incapable of forming a paddlewheel structure. This is largely due to the preference of Cu(I) to remain in a 2-fold coordinated environment. In both $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_2$ and $\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2$, the third ligand around a Cu ion is bound much more weakly and the formation of a fourth dative bond, e.g., in $\text{Cu}_2(\mu\text{-HCOO})_2(\mu\text{-HCOOH})_2$, is outright unfavorable.

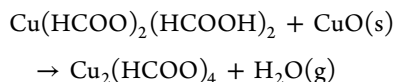
3.4.3. Predicted Thermodynamics under Experimental Conditions. Although the reactions are shown in a particular sequence in Figure 6 for Cu(II) and Figure 7b for Cu(I), the

formation of these complexes need not proceed in the same manner presented. Therefore, Table 1 presents also the standard reaction free energies to form (from CuO, $(\text{HCOOH})_2$, and H_2O) the most stable monomers $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})_2$ and $\text{Cu}(\text{HCOO})_2(\text{HCOOH})_2$ (reactions Mm , $m = 1$ and 2) and dimers $\text{Cu}_2(\text{HCOO})_4$, $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_4$, and $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_3(\text{OH})$ (reactions Dm , $m = 1, 2$, and 3). Table 1 also presents the standard reaction free energies to form (from Cu_2O , $(\text{HCOOH})_2$, and H_2O) the most stable monomers $\text{Cu}(\text{H}_2\text{O})_2(\text{HCOO})$ and $\text{Cu}(\text{HCOO})(\text{HCOOH})$ (reactions Mm' , $m = 1$ and 2) and dimers $\text{Cu}(\text{HCOO})_2$, $\text{Cu}_2(\text{H}_2\text{O})_2(\text{HCOO})_2$, and $\text{Cu}_2(\text{HCOO})_2(\text{HCOOH})_2$ (reactions Dm' , $m = 1, 2$, and 3). Henceforth, the free energies of these reactions are referred to as “formation” free energies (in contrast to the regular definition of a formation free energy, which uses the pure standard forms of the constituent elements as references). These standard formation Gibbs free energies then were extrapolated to experimental pressures of the gaseous reactants and products.

To calculate the partial pressure of $(\text{HCOOH})_2$ given the total pressure of all the formic acid species, the calculated dimerization free energy for $\text{HCOOH}(\text{g})$ (Supporting Information, methods section II) was used. The total chamber pressure is the result of unreacted HCOOH and the carrier gas N_2 (both deemed to be the majority gases), the product Cu complex, and H_2O (initially present or as a product of the neutralization reaction). 8.0×10^{-2} Torr or 1/10 of the vacuum base pressure (0.8 Torr) was used as the upper bound for the complex (vacuum condition is achieved pre-etching).

As for the H₂O vapor pressure, two extreme cases were defined: (1) 356 Torr and (2) 1/100 of the vacuum chamber base pressure, 8.0×10^{-3} Torr. Case (1) occurs when the vacuum condition is broken and never recovered (vacuum is broken between O₂ plasma exposure and HCOOH etching, and the saturation pressure of water at the experimental temperature of 353 K (80 °C) is 356 Torr).⁷⁷ Case (2) is relevant during etching (performed at 250 Torr) where the vacuum condition is recovered prior (base pressure of 0.8 Torr) and the chamber is subsequently filled with HCOOH/N₂ at 250 Torr, displacing more H₂O out of the reaction vessel ("dry" condition). The latter is the more relevant case in the current experiment, whereas the former is shown for comparison as a hypothetical case. The partial-pressure dependence of the reaction free energies is described in [Supporting Information, methods section III](#). Below, we predict that due to the difference in water content in the complexes described above, the monomer and dimers may prevail at the higher water vapor pressure conditions. For Cu(II), under water condition (1), Cu(H₂O)₂(HCOO)₂ is the most preferred monomer while Cu₂(μ-HCOO)₄ is favored among the three dimers, and overall. Under water condition (2) or the "dry" condition, the dehydrated Cu₂(μ-HCOO)₄ dominates and the monomeric Cu(HCOO)₂(HCOOH)₂ is favored over Cu(H₂O)₂(HCOO)₂, where the latter is significantly destabilized. For Cu(I), Cu₂(μ-HCOO)₂ prevails in both water conditions and is more difficult to form than Cu₂(μ-HCOO)₄ from their respective oxides of the same Cu oxidation state.

[Figure 8a](#) shows the reaction free energies for reactions M1, M2 and D1–D3 ([Table 1](#)) as a function of the total pressure of all the formic acid species ((HCOOH)_n), which can be directly measured or estimated from the experiment. Given a total etchant ((HCOOH)_n+N₂) pressure of 250 Torr, an upper bound of 250 Torr and a lower bound 125 Torr were defined for the partial pressure of (HCOOH)_n. These pressures are marked in the plots in [Figure 8](#). It becomes apparent from the plots that the monomer Cu(H₂O)₂(HCOO)₂ and dimers are most likely to form competitively during etching in the presence of ambient water vapor, albeit with Cu₂(μ-HCOO)₄ slightly more favored with a slightly positive formation free energy of 0.32–0.34 eV/Cu at relevant (HCOOH)_n pressures. However, under the "dry" condition, only the water-free dimer Cu₂(μ-HCOO)₄ prevails, with nearly zero formation free energy that is between −0.01 and +0.01 eV/Cu. Thus, recovery of the vacuum prior to etching is imperative to drive the etching to spontaneity, because it deprives the chamber of one of the neutralization products: water vapor. The formation of Cu(H₂O)₂(HCOO)₂ under this condition is highly unfavorable; however, the thermodynamics of Cu-(HCOO)₂(HCOOH)₂ formation is enhanced. Cu-(HCOO)₂(HCOOH)₂ then may act as a more favorable intermediate for the formation of Cu₂(μ-HCOO)₄. The dimerization, which releases (HCOOH)₂, is between −0.07 and −0.05 eV/Cu from 125 to 250 Torr total formic acid at 353 K (these energies are independent of the partial pressure of H₂O). Cu₂(μ-HCOO)₄ can also form from the reaction of Cu(HCOO)₂(HCOOH)₂ with a unit of CuO(s):



In such a reaction, the two protons from the two HCOOH ligands react with a lattice O to form H₂O, and the resulting four free carboxyl O anions anchor onto and extract a Cu(II) ion. This pathway is also spontaneous at 0.008 Torr H₂O, with a reaction free energy of −0.06 eV/dimer or −0.03 eV/Cu at 353 K (independent of the partial pressure of (HCOOH)₂) and is thus competitive with the dimerization reaction described above. Note that this step is unfavorable at 356 Torr H₂O (+0.27 eV at 353 K).

Because the experiment demonstrated the existence of Cu(OH)₂, the same plots using Cu(OH)₂ instead of CuO as reactant were obtained. This is easily achieved by subtracting the water-vapor-pressure-dependent free energy to form Cu(OH)₂ from CuO and H₂O(g) (reaction j in [Figure 6](#) and [Table 1](#)) from the free energies in [Figure 8a,b](#). The resulting plots appear in [Figure 8c,d](#). Because j has a positive free energy (especially at very low H₂O partial pressures), the formation energies are more negative than in the case of CuO. Despite the energy shift, the relative stability of the complexes is maintained. However, it is known experimentally that under low humidity conditions for oxidation, little or no Cu(OH)₂ forms. The relevant reactions are therefore those that involve CuO as the reactant.

As for Cu(I) complexation, [Figure 8e,f](#) depicts the formation free energy versus total formic acid pressure of the notable monomer and dimer Cu(I) complexes: M1', M2' and D1'–D3' ([Table 1](#)). In both water-rich and -deprived conditions, the dimer Cu₂(μ-HCOO)₂ (D1') is the most favorable to form. Another dimer, Cu₂(μ-HCOO)₂(HCOOH)₂ (D3'), is the second most favorable species. The free energy of formation of Cu₂(μ-HCOO)₂ from Cu₂O ranges from 0.33 to 0.34 eV/Cu at 356 Torr H₂O and 353 K, which slightly improves to 0.17–0.18 eV/Cu at 0.008 Torr H₂O. Addition of two HCOOH to form D3' slightly increases the free energy of this dimer by ~0.1 eV/Cu. Cu(HCOO)(HCOOH) (M2') on the other hand is ~0.2 eV/Cu higher in energy than D1'. Cu(HCOO)(HCOOH), therefore, cannot facilitate the formation of the most stable dimer.

Etching of Cu(I) through complexation is thermodynamically less favorable than Cu(II) by ~0.2 eV/Cu, and unlike in Cu(II), a Cu(I) monomer that is energetically adjacent to the most stable Cu(I) dimer does not exist. Therefore, Cu(II) is expected to be the species that may be etched by HCOOH, which highlights the importance of the oxidation half-cycle with plasma O₂, to oxidize the newly exposed Cu after etching and any Cu(I) that may form.

The etching of oxidized Cu mediated by HCOOH vapor may be regarded as rather "inefficient" considering it proceeds only a few nm per cycle (2.4 nm/cycle, [Figure 4a](#)) with a 5 min exposure to HCOOH per cycle, but desirable because it enables a nanometer-resolved etch rate. This "inefficiency" is reflected in the predicted nearly zero etching free energy of Cu(II) through complexation under experimental conditions ([Figure 8b](#)). Atomic resolution thus is achieved by the complexation reactions that are at or near equilibrium.

3.4.4. Possible Avenues for Product Detection. The fate of Cu₂(μ-HCOO)₄ is unknown in the experiment. It is likely that it may collect on the colder part of the vacuum system, perhaps forming precipitates. Cu(II), formate, and water are known to form a monoclinic copper formate tetrahydrate crystal: Cu(HCOO)₂·(H₂O)₄, composed of alternating layers of Cu(HCOO)₂ and (H₂O)₄ along its *c*-axis.^{78,79} It is thus more likely to form this compound at higher water vapor

conditions at lower temperatures, which perhaps can be an avenue for future experimental detection of the etching products.

4. SUMMARY AND CONCLUSIONS

Atomic layer etching of Cu was achieved by using an O₂ plasma as the controlled oxidant and HCOOH vapor as the self-limiting etchant. This procedure was cycled until the desired amount of Cu was etched. From first-principles DFT +*U* with van der Waals correction simulations, it was shown that CuO can be etched by HCOOH, by first neutralizing the oxide and then forming a complex with the HCOO[−] ions. Under relevant experimental pressures (125 to 250 Torr formic acid etchant, with the partial pressures of the complex and water 1/10 and 1/100 of the vacuum base pressure of the reactor, i.e., 0.8 Torr, respectively), the DFT analysis predicted that the water-free dimeric gas-phase product Cu₂(μ-HCOO)₄ is most likely to form under water-deprived conditions, highlighting that re-establishing vacuum prior to acid etching is critical. The formation of Cu₂O from CuO, where HCOOH acts as a reductant, was also considered as a reaction prior to complexation (reductive etching). However, Cu(I) etching from Cu₂O was found to be less thermodynamically favorable than Cu(II) etching from CuO. To conclude, etching the oxide presents a clear advantage over directly etching metals. The use of vapor phase over solution phase of the acidic etchant also provided control over the etching selectivity and anisotropy.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.0c08932>.

Supplementary methods, optimized crystal structures and phonon densities of states of bulk CuO, Cu(OH)₂, and Cu₂O, SEM images of as-deposited Cu, after oxidation, and after HCOOH etching at different etching times, Cu-LMM Auger line XPS and Wagner plot for the different Cu species, HCOOH monomer and dimer structures, alternative structures of some of the monomeric Cu(II)- and Cu(I)-(formate, formic acid) complexes, structures and free energies of formation of the Cu(II)-aquahydroxo complexes relative to Cu(OH)₂ and H₂O(g), data tables of the optimized lattice parameters, *k*-point sampling, and relative energies of the different magnetic ordering for CuO and Cu(OH)₂, data tables summarizing the enthalpic and entropic energies of the solids (vibrational) and gaseous molecules (vibrational, rotational, and translational), data table showing the relative energies of the different magnetic orderings for the Cu(II) dimer complexes (PDF)

Optimized-atomic-structure files of the ground-state bulk crystals (VASP format) and isolated molecules (xyz format) (ZIP)

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Notes

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