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3 **Low porosity, high areal-capacity Prussian blue analogue**

4 **electrodes enhance salt removal and thermodynamic**

5 **efficiency in symmetric Faradaic deionization with**

6 **automated fluid control**

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26 **Abstract**

27 Prussian blue analogues (PBAs) show great potential for low-energy Faradaic
28 deionization (FDI) with reversible Na-ion capacity exceeding 4 mol/L in the solid-state.
29 However, past continuous-flow demonstrations using PBAs in FDI were unable to
30 desalinate brackish water to potable levels using single-pass architectures. Here, we
31 show that recirculation of effluent from a symmetric cation intercalation desalination cell
32 into brine/diluate reservoirs enables salt removal exceeding 80% at thermodynamic
33 efficiency as high as 80% when cycled with 100 mM NaCl influent and when controlled
34 by a low-volume, automated fluid circuit. This exceptional performance is achieved using
35 a novel heated, alkaline wet phase inversion process that modulates colloidal forces to
36 increase carbon black aggregation within electrode slurries to solidify crack-free, high
37 areal-capacity PBA electrodes that are calendered to minimize cell impedance and
38 electrode porosity. The results obtained demonstrate the need for co-design of auxiliary
39 fluid-control systems together with electrode materials to advance FDI beyond brackish
40 salinity.

41

42 **1.0 Introduction**

43 With more than half of humanity presently facing freshwater scarcity for one month
44 during a given year (Mekonnen and Hoekstra, 2016), desalination of sea and brackish
45 waters could increase freshwater access (Elimelech and Phillip, 2011). However, recent
46 analysis estimates that roughly 1.5 units of concentrated brine are generated globally for
47 every unit of desalinated water, averaged across all water sources, sectors, and
48 technologies (Jones et al., 2019). Depletion of natural freshwater sources motivates the
49 need for new water treatment technologies while also combatting climate change by
50 consuming less energy and replacing aging infrastructure (Mauter and Fiske, 2020).
51 While reverse osmosis (RO) presently comprises the majority of global desalination
52 capacity (Jones et al., 2019), large RO plants are required to achieve high energy
53 efficiency and RO is limited to brine concentrations lower than 1.6 mol/L as monovalent
54 salt, assuming an 80 bar limit for spiral wound RO pressure vessels (Subramani and
55 Jacangelo, 2014). Electrically driven desalination processes avoid the mechanical
56 limitations of RO by separating dissolved ions rather than H₂O molecules. Among such
57 processes (Suss et al., 2015; Tang et al., 2019), Faradaic deionization (FDI) using cation
58 intercalation materials (CIMs) shows great promise on the basis of reversible Na
59 concentrations in the solid-state (namely 5.0 mol/L for the Prussian blue analogue
60 investigated here with 65 mAh/g charge capacity and 2 g/cc nominal density) exceeding
61 seawater salinity (0.5 mol/L as NaCl) by ten-fold. Techno-economic analysis of FDI and
62 comparison with the more frequently studied capacitive deionization (CDI) shows the
63 former method outputs more desalinated water at lower cost while being a more compact
64 technology, in part due to CIMs having greater capacity (Metzger et al., 2020). Despite

65 our early modeling that introduced a symmetric Na-ion battery architecture to desalinate
66 seawater using CIMs (Liu and Smith, 2018; Smith, 2017; Smith and Dmello, 2016),
67 subsequent experimental demonstrations thereof failed to realize more than 30% salt
68 removal from continuously flowing brackish water (Kim et al., 2017; Porada et al., 2017;
69 Reale et al., 2019). We posit here that efficiency losses within auxiliary systems must be
70 understood in conjunction with electrode-scale transport processes to extend FDI using
71 CIMs beyond brackish salinity.

72 The configuration of electrolyte flow through porous electrodes (Liu and Smith,
73 2018; Reale et al., 2019; Smith, 2017; Son et al., 2020) and charge transport within
74 porous electrodes (Reale et al., 2019; Shrivastava and Smith, 2018) are known to affect
75 rate capability and energy consumption in FDI using symmetric CIM electrodes, but past
76 results suggest that unknown mechanisms remain and prevent the salt-removal potential
77 of CIMs from being accessed. In FDI, saltwater can flow through a gap between
78 electrodes in a flow-by configuration or directly through the porous structure of electrodes
79 sandwiched within a cell in a flow-through configuration. Flow-through electrodes
80 incorporating Prussian blue analogue (PBA) CIMs, which we first demonstrated (Reale et
81 al., 2019), have shown improved salt removal (Smith, 2017), charge utilization (Liu and
82 Smith, 2018; Smith, 2017; Son et al., 2020), and specific energy consumption (Smith,
83 2017; Son et al., 2020) compared to flow-by PBA electrodes due to charge-efficiency loss
84 at electrode/channel interfaces in flow-by (Liu and Smith, 2018; Smith, 2017) and smaller
85 ohmic drop in flow-through (Son et al., 2020). Further, our experiments introducing flow-
86 through FDI achieved rapid salt removal with one order of magnitude lower specific
87 energy consumption, as provided by judicious choice of electronically conductive

88 additives (Reale et al., 2019) and inspired by our earlier work demonstrating that
89 nanoparticle agglomeration limits transport in electrodes containing PBAs (Shrivastava
90 and Smith, 2018). However, charge efficiency diminished at higher current densities due
91 to low utilization shortening cycle time, an effect that complementary modeling indicated
92 was likely a result of strong salt concentration gradients in both the streamwise and
93 transverse directions within porous electrodes (Reale et al., 2019).

94 Along those lines, our past modeling revealed that streamwise salt concentration
95 gradients in feedwater result in streamwise state-of-charge gradients in CIMs within the
96 porous electrodes through which feedwater flows (Liu and Smith, 2018). While such
97 state-of-charge gradients lead to efficiency loss and decreased CIM capacity utilization,
98 we predicted that recirculation of feedwater can minimize such deleterious effects (Liu
99 and Smith, 2018). While recirculation eliminates the coupling between flow rate and salt
100 removal which constrains single-pass architectures, past studies (Liu et al., 2021;
101 Pothanamkandathil et al., 2020; Singh et al., 2020) provided limited detail concerning the
102 approaches used to repeatedly cycle feedwater through a single two-electrode FDI cell,
103 and such studies only demonstrated a cumulative salt removal as high as 40 mM (Singh
104 et al., 2020). Along similar lines, others have flowed feedwater in series through flow
105 compartments in an electrodialysis (ED) stack, while only reducing the salinity of 25 mM
106 influent by 30% (Kim et al., 2017). The development of novel high-capacity electrode
107 materials for electrochemical desalination including CDI and ED has been regarded as
108 unable to significantly reduce energy consumption compared to RO (Patel et al., 2020).
109 However, we show here that improvements can be achieved in FDI through reduction of
110 concentration gradients, increasing electrode conductivity, and development of auxiliary

111 flow-control systems optimizing salt removal in conjunction. Accordingly, we demonstrate
112 and detail for the first time an automated multi-pass FDI architecture using recirculation
113 from brine and diluate reservoirs enabled by custom, low-volume valves and a low-cost
114 Arduino control system.

115 Beyond such effects, we postulate that the porosity of flow-through FDI electrodes
116 can be engineered to increase degree-of-desalination in flow-through FDI cells. Mass
117 balance on a representative volume element of a porous electrode with porosity ε and
118 CIM volumetric loading ν suggests that decreasing porosity from as-cast values of 85%
119 (Reale et al., 2019) to 67% could increase degree-of-desalination by over three-fold,[†]
120 which is conceivable by calendering electrodes. Calendering, the process of compressing
121 electrodes between two rollers, is known to increase electronic conductivity and improve
122 rate capability (van Bommel and Divigalpitiya, 2012), but decreased porosity and
123 increased tortuosity are known to reduce ionic conductivity (Ebner et al., 2014;
124 Landesfeind et al., 2016; Malifarge et al., 2017). While these competing effects may result
125 in a certain degree of calendering that maximizes FDI performance, the fabrication of
126 thick electrodes with the areal capacity levels needed to create low-porosity electrodes
127 using calendering is a challenge in itself. The reduction of cost and inactive mass in Li-

[†] Neglecting volume within electrodes comprised by conductive additives and binder and assuming perfect transmission of solution through porous electrodes, the maximum cation concentration change within solution $(\Delta c)_{soln}$ that could be effected by transferring such cations into a CIM with a reversible cation concentration $(\Delta c)_{solid}$ is given by $(\Delta c)_{soln}/(\Delta c)_{solid} = (1 - \varepsilon)/\varepsilon$.

128 ion batteries (Gallagher et al., 2016) has motivated development of alternative electrode
129 fabrication processes to yield high areal capacity (Zolin et al., 2019) due to the tendency
130 of conventionally-prepared electrodes to crack during drying after slurry casting. To this
131 end, we use a wet-phase inversion (WPI) process that eliminates the capillary-bridge
132 induced cracking that arises during solidification by drying. While we (Bhat et al., 2019;
133 Reale et al., 2019; Reale and Smith, 2018) and others (Lim et al., 2009) have used WPI
134 to fabricate electrodes for FDI and capacitive deionization using a room temperature pure
135 water bath, here we introduce a heated, alkaline WPI process inspired by the Derjaguin-
136 Landau-Verwey-Overbeek (DLVO) theory of inter-particle forces in colloids (Goodwin,
137 2009). The DLVO theory has previously predicted percolation thresholds in simulated
138 dispersions (Maria et al., 2015) and is applied here towards slurry constituents to induce
139 solidification of thick, high-conductivity electrodes that we subsequently calender. While
140 past simulations based on Brownian dynamics of conductive particles in electrode slurries
141 predicted that larger particles form conductive paths more easily (Cerbelaud et al., 2015),
142 our previous experiments (Reale et al., 2019) on slurries containing PBA nanoparticles
143 and carbon black showed trends opposite those simulations. Rather than expecting
144 particles to pack together randomly as in Brownian dynamics, we posit per DLVO theory
145 that van der Waals and electrostatic forces influence their dynamics to a degree
146 dependent on particle radii, zeta potential, and Hamaker constants of the associated
147 particles and on the dielectric properties of intervening fluid. While the interplay between
148 such forces has been simulated for Li-ion battery electrodes, showing variations in
149 electrode microstructure dependent on the mass fraction, particle size, and temperature
150 of both CIM and additive particles (Zhu et al., 2014, 2011), few studies (Cho et al., 2007)

151 have previously exploited understanding of such interactions in experimental electrode
152 fabrication. In doing so, our colloid-inspired electrode fabrication process results in more
153 electronically conductive and mechanically stable electrodes that do not crack at high cast
154 thicknesses. Calendering of these electrodes increases their density and electronic
155 conductivity, which is shown to enhance salt removal and thermodynamic efficiency in
156 FDI.

157 A brief outline of this article follows. We first discuss a novel wet phase inversion
158 process and the effects of electrode calendering, which we use in conjunction to fabricate
159 electrodes with low porosity, high areal capacity, and high rate capability. We then
160 present results for the cycling of such electrodes in FDI desalination experiments. In such
161 experiments we employ a custom lab-scale recirculating FDI cell to alternate flow paths
162 such that feedwater in one reservoir continuously desalinates while another reservoir
163 simultaneously concentrates. After several cycles using the densest electrodes, the
164 dilute reservoir reached 90% salt removal from 100 mM NaCl influent before reaching a
165 limit caused by desalinated and concentrated effluent mixing reducing charge efficiency.
166 The use of denser electrodes is further shown to decrease energy consumption,
167 achieving 80% thermodynamic energy efficiency (TEE) at 80% salt removal.

168

169 **2. Materials and Methods**

170 *2.1 Electrode fabrication, conductivity measurements, and flooded cell characterization:*

171 PBA nanoparticles and associated slurries were prepared as in our previous work
172 (Reale et al., 2019; Shrivastava and Smith, 2018) with electrodes composed of 85 wt%
173 nickel hexacyanoferrate (NiHCF) PBA, 5 wt% EC-600JD Ketjen black, and 10 wt% Solef

174 polyvinylidene fluoride (PVDF) binder. The first electrodes were cast at 150 μm thick onto
175 100 μm thick 2010A high-purity graphite foil current collector (Ceramaterials), and
176 subsequent electrodes were cast at higher thicknesses of 250 μm and 320 μm as
177 described in the Results and Discussion Section. After being cast and solidified using
178 wet phase inversion (WPI, as described subsequently), the electrodes were calendered
179 using an MTI roll press to a targeted thickness of 150 μm , as measured by an MTI
180 thickness gauge. Porosity was then calculated from the densities of the total electrode
181 and its solid constituents, $\varepsilon = 1 - \rho_{\text{electrode}}/\rho_{\text{solid}}$. Electrochemical cycling of these
182 electrodes was performed at several C-rates in a 1M NaCl flooded cell versus a Ag/AgCl
183 reference electrode with a graphite counter electrode.

184 We used WPI to solidify electrodes after casting. Here, an as-cast slurry film
185 supported on a current collector is immersed in a water bath to displace n-methyl-2-
186 pyrrolidone (NMP) solvent and solidify the PVDF binder dissolved therein to form a
187 mechanically stable electrode (Bhat et al., 2019; Reale et al., 2019; Reale and Smith,
188 2018). While previous WPI processes used room temperature deionized water baths for
189 phase inversion, we explored modified bath conditions to improve electrode properties.
190 Initial tests were done using 1 μm Al_2O_3 particles (AdValue Technology) as a surrogate
191 for PBA material due to supply limitations for the latter material, as in our previous work
192 optimizing electrode conductivity without calendering (Reale et al., 2019). As Timcal
193 Super C45 carbon black (MTI) had previously been shown to be an inferior conductive
194 additive compared to Ketjen black, C45 was used in initial studies with the expectation
195 that improvements in conductivity would be more obvious than with Ketjen black. To
196 measure the electronic conductivity of electrodes, slurries were cast onto an insulative

197 PTFE substrate, and conductivity was measured using an in-house built four-point probe
198 (Reale et al., 2019).

199 To influence the inter-particle forces in slurries, a 200 mL WPI water bath was used
200 to explore the effects of three different experimental parameters: (1) using a hot plate to
201 heat the water between 25°C and 85°C, (2) increasing bath pH from 7 to 12 by adding
202 NaOH, and (3) increasing salt concentration by dissolving NaCl up to 450 mM (see SI
203 Section S3). PBA electrodes immersed in the high pH bath also displayed structural
204 stability when cast at high thickness, solidifying at 320 μ m when an electrode that thick
205 would normally crack during drying if phase inverted using a room temperature, neutral
206 pH water bath.

207

208 *2.2 FDI system design and control:*

209 The custom built FDI cell used in our previous study (Reale et al., 2019) was
210 employed for desalination experiments in conjunction with a custom recirculating fluid
211 circuit built around it (see Fig. 4, SI Section S1, and Video S1). The FDI cell used two
212 PBA electrodes separated by a Neosepta AMX anion exchange membrane (Astom Corp.).
213 In all desalination experiments two test tubes were filled with 5 mL of saltwater to be
214 converted to brine and dilute during cycling, with initial concentrations of 100 mM, or 200
215 mM in a separate experiment. To direct flow, an Arduino controller actuated four servos
216 to pinch branching paths of rubber tubing, ensuring that feedwater from only one reservoir
217 flowed through either side of the FDI cell (Video S1). Flow rates of both effluent streams
218 were kept constant and equal by using a Masterflex peristaltic pump, and all experiments
219 used a flow rate of 1 mL/min. This flow rate is 4x higher than our previous experiments

220 with a single-pass system (Reale et al., 2019) to take advantage of the recirculation,
221 keeping concentration gradients low. With the diluate and brine flow rates being equal,
222 the water recovery rate in all experiments was 50%. Images of the complete cell design,
223 details of its fluid circuit, and its electrical control system are found in SI Section S1. Salt
224 concentration at the two inlets and two outlets was measured using four custom-built flow-
225 through conductivity sensors connected to an eDAQ isoPod which recorded conductivity
226 using the program PodVu.

227

228 *2.3 Desalination experiments*

229 All desalination experiments used a current density of 1 mA/cm² applied to 2.5 cm²
230 electrodes by a Biologic VMP-3 potentiostat over a voltage window of +/- 0.45V. Based
231 on our measured values of hydraulic permeability, we estimate a pressure drop across
232 the FDI cell of 71.6 kPa and 430 kPa, respectively, for cycling with the least and highest
233 areal-capacity electrodes tested here at a 1 mL/min flow rate. The time delay between
234 OC starting and IV actuation was set to 20 seconds in all FDI experiments, roughly twice
235 the duration of the time delay between IV and OV actuation, to uniformize concentration
236 on the respective sides of the cell. During the OC period before IV actuation,
237 electrochemical impedance spectroscopy (EIS) tests were performed with frequencies
238 ranging from 100 kHz to 100 mHz to study how impedance changes from cycle to cycle.
239 Lower frequencies were not tested because doing so would prolong experiments. The
240 time delay between actuation of IVs and OVs was different for each electrode pair and
241 was calculated using a calibration process described in SI Section S2. Prior to each
242 experiment, the conductivity probes used were calibrated with 100 and 200 mM influent,

243 known to respectively have conductivities of 1.067 and 2.034 S/m at 298K (Chambers et
244 al., 1956).

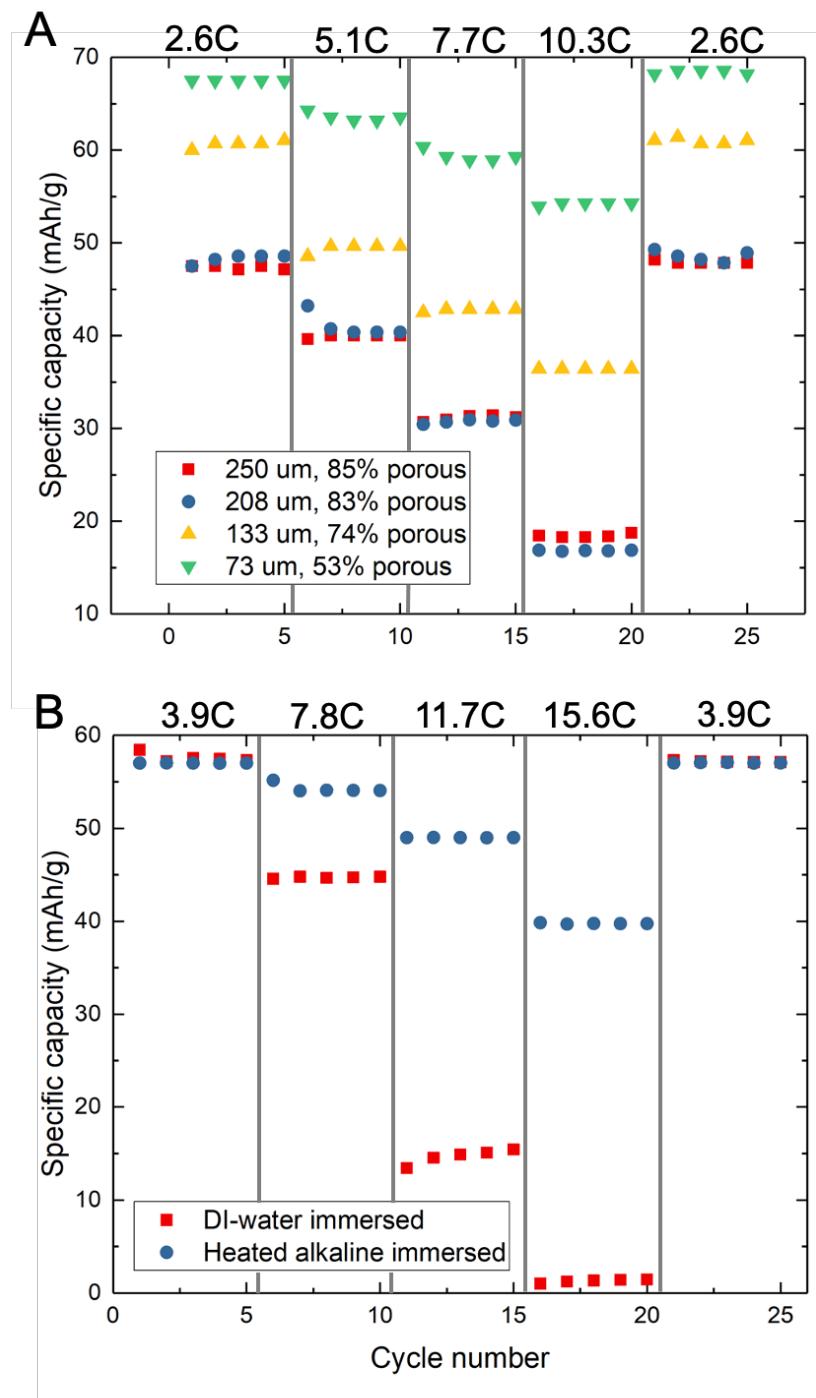
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246 **3.0 Results and Discussion**

247 *3.1 Electrode fabrication techniques*

248 We first report on experiments aimed at fabricating high conductivity, high areal-
249 capacity PBA electrodes, predicated on the hypothesis that their use in FDI will increase
250 salt removal. To fabricate PBA electrodes with high areal loading and low porosity we
251 initially used our past WPI procedure (Reale et al., 2019) (using a room temperature
252 deionized (DI) water bath for phase inversion) to fabricate 250 μm thick electrodes that
253 were subsequently calendered to lower thicknesses, yielding electrodes of varied
254 porosity. Figure 1A shows the results for cycling of an electrode with different degrees of
255 calendering using a 1M NaCl flooded cell with a graphite counter electrode and a Ag/AgCl
256 reference electrode. These results indicate that as electrode thickness and porosity
257 decrease, electrode rate capability increases. Between rates of 2.6C and 10.3C,
258 utilization of an uncalendered 250 μm electrode reduced by 64%, while the same
259 electrode calendered down to 72 μm showed only an 18% reduction in utilization and
260 34% greater capacity at 2.6C relative to 250 μm electrodes at the same rate. Returning
261 to the lowest C-rate after cycling each sample at 10.3C confirmed that the capacity drops
262 observed at high rate were not due to electrode degradation. Despite the concern of low
263 porosity increasing ionic resistance in porous electrodes (Ebner et al., 2014; Landesfeind
264 et al., 2016), calendered electrodes showed no rate capability decrease due to lost pore
265 volume. However, we found that electrodes cast at thickness greater than 250 μm readily

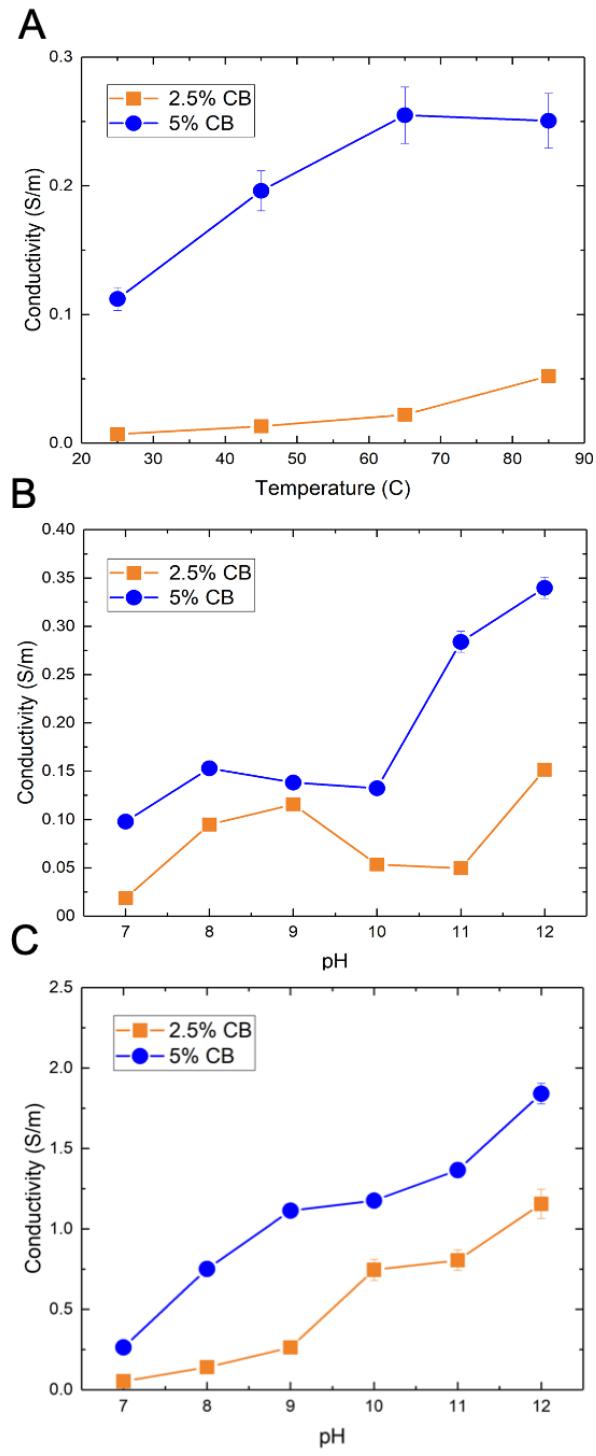
266 cracked when using conventional WPI, motivating our search for improved electrode
267 fabrication conditions.



268
269 **Figure 1:** (A) Specific capacity for a PBA electrode fabricated with conventional WPI at
270 different degrees of calendering at different C-rates. (B) Capacities of uncalendered

271 electrodes fabricated with conventional WPI and 85°C, pH 12 water immersed electrodes.
272 All cycling experiments were performed in a 1M NaCl flooded cell.
273

274 To overcome areal loading limitations and increase electrode performance overall,
275 we analyzed the colloidal interactions between particles in electrode slurry occurring
276 during WPI processes. We postulate that the forces between active (PBA) and
277 conductive (carbon black) particles during microstructure formation of an electrode can
278 be engineered to promote aggregation of conductive particles and increase effective
279 electronic conductivity. To guide our investigation of such we employ DLVO theory, the
280 application of which suggests that decreased conductor-active particle aggregation is
281 achievable by conductors and active particles having like-signed zeta potential, making
282 them mutually repulsive (see SI Section S3). Furthermore, DLVO theory suggests that
283 temperature affects electrostatic interactions. Subsequently, we alter WPI bath conditions
284 and show that repulsion between carbon and active particles can be increased, causing
285 more carbon to aggregate into a conductive backbone and yielding increased electronic
286 conductivity.



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Figure 2: Electronic conductivity for electrodes with 2.5 and 5 wt% C45 carbon black and surrogate active particles (Al_2O_3) made using (A) increasing bath temperature and (B) increasing bath pH. Error bars are found from the 5 μm variability in electrode thickness which determines how conductivity is calculated using a four-point probe. (C) Electronic conductivity for electrodes made at 85°C using an alkaline bath.

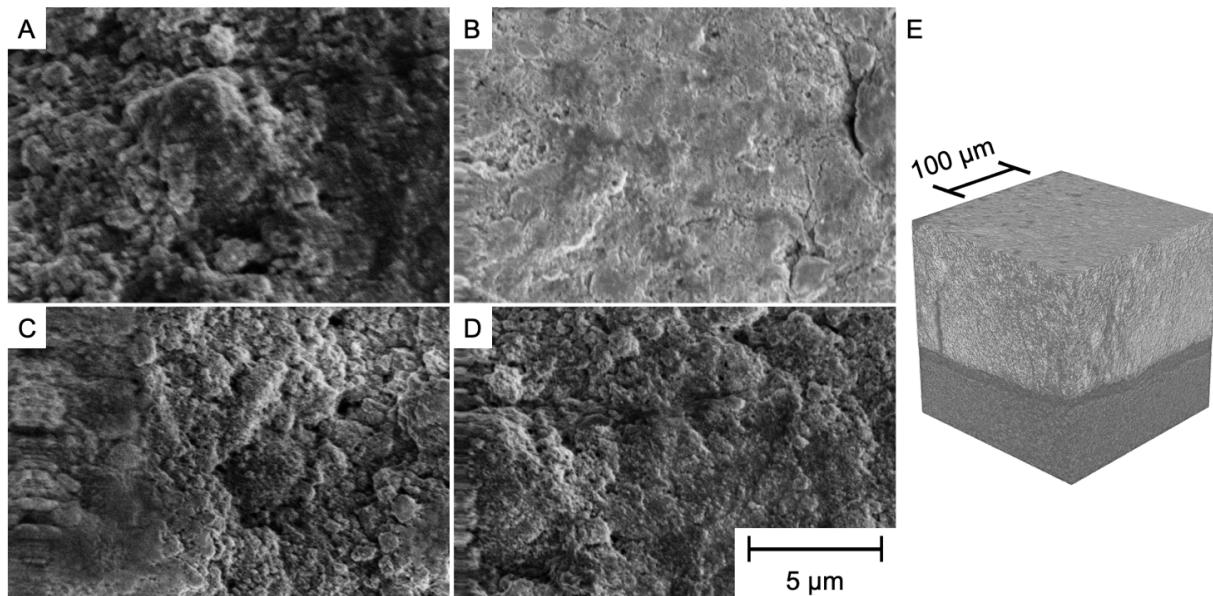
294 Accordingly, we performed modified WPI experiments on cast electrode slurries to
295 characterize the effects of the temperature and pH of inversion baths on electronic
296 conductivity, so as to modulate electrostatic colloidal interactions respectively via Debye
297 length and zeta potential. To do this we formulated slurries comprised of conductive
298 particles (C45 carbon black) and surrogate particles replacing low-conductivity PBA
299 (Al_2O_3). Figures 2A and 2B show variations of electronic conductivity respectively with
300 temperature and pH, where temperatures of 80-90°C and pH 12 are shown to produce 2-
301 3 times higher conductivity than the baseline case at room temperature and neutral pH
302 when 5 wt.% carbon black is used. The changes in electronic conductivity observed
303 among the different WPI conditions, despite containing the same constituent mass
304 fractions, are qualitatively consistent with DLVO theory. Increasing pH causes zeta
305 potential to decline for both particle phases, and a sufficient increase thereof can cause
306 surrogate active particles to become negatively charged (Singh et al., 2005). However,
307 carbon black particles have been found to possess negative zeta potential at pH 7 (Sis
308 and Birinci, 2017), indicating that surrogate active particles and carbon black experience
309 strong electrostatic attraction at pH 7. Because increasing pH beyond 7 causes carbon
310 black particles to have increasingly negative zeta potential, carbon and surrogate active
311 particles repel each other in alkaline solution. Additionally, increasing bath temperature
312 gives carbon particles more kinetic energy to overcome potential energy barriers,
313 increasing the likelihood of carbon particle aggregation. Variations of the corresponding
314 potential energy curves are presented in the SI to illustrate these concepts (Section S3).
315

316 Because the third condition for the WPI bath, dissolution of inert salt, produced
317 mild effects on electrode formulation (see SI Section S3), we next explored the potential
318 for synergistic interactions of increased temperature and pH to enhance electrode
319 formulation. Figure 2C shows that the resulting electronic conductivity, obtained for a
320 bath heated at 85°C using different levels of alkaline pH, increases to a greater degree
321 than when using modified temperature or pH conditions separately. In electrodes
322 containing 2.5 wt% carbon black, conductivity increased by up to 167 times compared to
323 electrodes immersed in room-temperature DI water.

324 Informed by findings obtained with surrogate active material and C45 carbon
325 black, we subsequently used WPI bath conditions that yielded highest electronic
326 conductivity (85°C and pH 12) to fabricate PBA electrodes containing Ketjen black
327 conductor, for which our previous work (Reale et al., 2019) demonstrated improved
328 electronic conductivity relative to C45 carbon black. Flooded cell cycling comparing WPI
329 processes was performed, shown in Fig. 1B. The electrode wet-phase inverted in a
330 heated, alkaline bath showed superior rate capability compared to electrodes prepared
331 using standard conditions, which is consistent with the results obtained with surrogate
332 material that showed increased electronic conductivity under such conditions. In addition,
333 the modified WPI conditions were shown to prevent cracking of electrodes with
334 thicknesses as large as 320 µm, despite an electrode immersed in 85°C, neutral pH water
335 cracking at 320 µm. This finding confirms the critical role of WPI bath pH in preserving
336 the thick electrodes' mechanical integrity. We postulate that binder-solution interactions,
337 such as the Hofmeister effect (Thormann, 2012), may also increase binder uniformity
338 within electrodes and consequently increase mechanical strength.

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340



341

342 **Figure 3:** SEM Images of a conventional PBA electrode (A) before calendering and (B)
 343 after calendering. Images (C) and (D) show a heated, alkaline-bathed electrode before
 344 and after calendering, respectively. All four images use the same scale bar. (E) X-ray CT
 345 image of a calendered PBA electrode (light gray) prepared using the modified WPI
 346 process supported on graphite foil (dark gray).

347

348 To acquire visual evidence for the effect of electrode fabrication conditions on their
 349 microstructure, SEM images of DI water-immersed and heated alkaline-immersed
 350 electrodes were obtained (Fig. 3), including images taken before and after calendering to
 351 50% of their initial thickness. While the microstructures of uncalendered electrodes
 352 prepared using conventional and modified WPI processes were indistinguishable (Figs.
 353 3A, 3C), WPI conditions were shown to affect the texture of calendered electrodes
 354 substantially (Fig. 3B and 3D). Specifically, the conventional WPI process produced a
 355 relatively smooth and occluded calendered surface compared with the modified WPI
 356 process, consistent with the postulate that increased pH and temperature during WPI
 357 increases binder uniformity throughout electrodes. This observation is also consistent

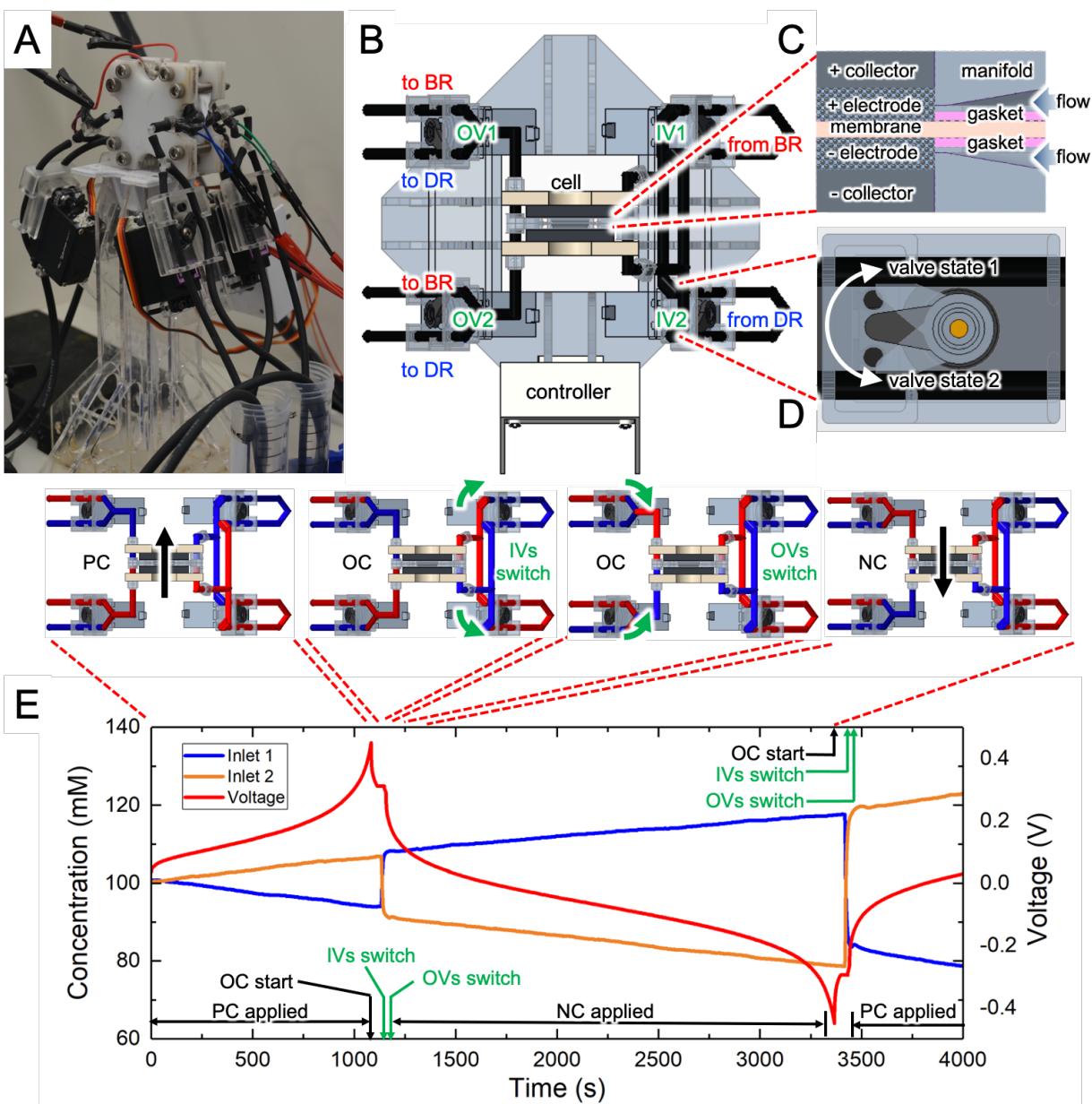
358 with the results of flooded-cell cycling which showed improved rate capability for
359 electrodes prepared using the modified WPI process. Fig. 3E shows an X-ray computed
360 tomography (CT) scan of a calendered electrode prepared by modified WPI, confirming
361 the process yields a homogeneous distribution of PBA particles within the microstructure
362 over a 200 μm by 200 μm area. However, a larger area of 620 μm by 790 μm reveals
363 defective fissures constituting approximately 5% to 10% of the overall volume in the
364 electrode (Fig. S11 and Video S2). To understand the impact of such fissures, we
365 measured these electrodes' hydraulic permeability using a simple gravity-driven
366 apparatus (Reale et al., 2019). The densest calendered electrodes had a permeability of
367 0.23 μm^2 compared with the uncalendered electrodes' 1.38 μm^2 due to their decreased
368 porosity and average micropore radius. This finding indicates that flow is forced through
369 the microporous regions of these electrodes and is therefore not "short-circuited" through
370 defective fissures. These results for electrode microstructure, flooded cell cycling, and
371 electronic conductivity motivated use of the modified WPI process and electrode
372 calendering with CIMs in a cation intercalation desalination FDI cell.

373

374 *3.2 Recirculating FDI desalination performance*

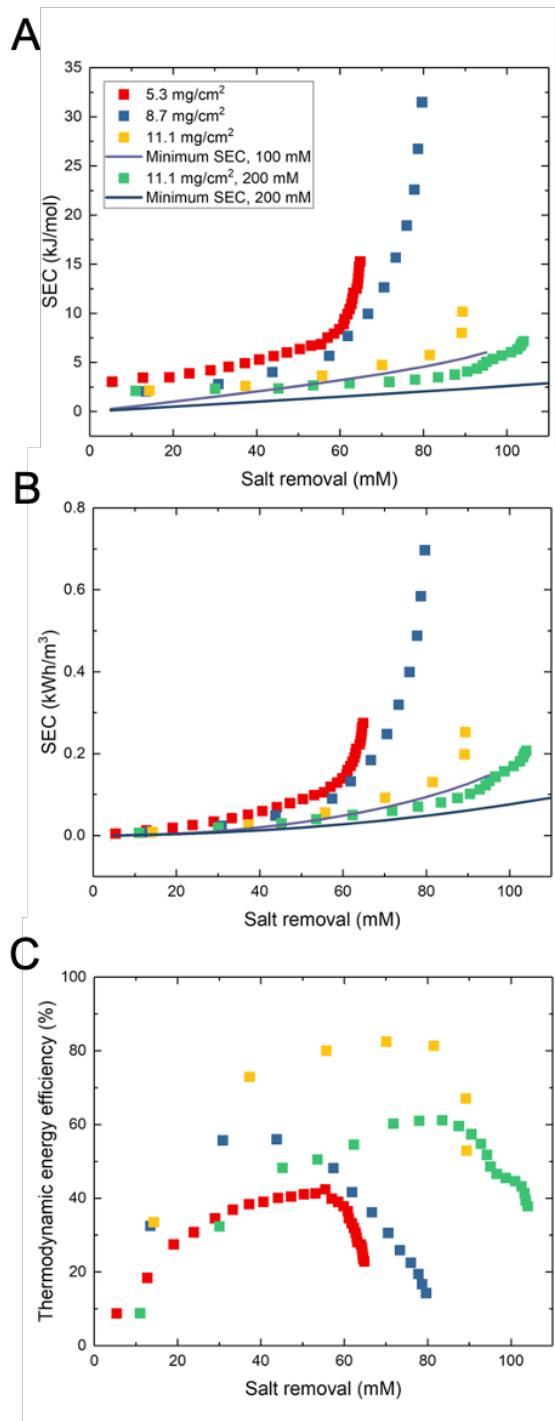
375 FDI experiments were conducted using an in-house custom-built flow cell and
376 integrated fluid recirculation system (Figs. 4A-4D), detailed in SI Section S1. Valves at
377 the outlets (OVs) and inlets (IVs) of both electrodes direct salt water respectively to and
378 from the reservoirs for brine (BR) or diluate (DR), as shown in Figs. 4B and 4C. Figure
379 4E shows the corresponding variations of cell voltage and influent salt concentrations with
380 time, the latter measured by conductivity probes at the inlet regions of both electrodes

381 when the cell is subjected to galvanostatic cycling with either positive (PC) or negative
382 (NC) current, followed by a brief open-circuit (OC) period. In addition to constructing the
383 cell with the lowest internal volume possible by using narrow inner diameter tubing and
384 custom parts dimensioned to minimize volume, experiments were conducted to optimize
385 valve timing to reduce mixing between diluate and brine streams. Pause periods were
386 added at the end of each PC/NC period, minimizing mixing by accounting for residence
387 time in the internal volume of the system, as described in Section S2 of the SI. Hence,
388 Figure 4E and Video S1 show a finite-time delay between switching events of electrical
389 current and both IVs and OVs, denoted as 'IVs switch' and 'OVs switch.' After calibrating
390 such events we observed salt concentration varying linearly with time during initial cycles
391 and that salt removal per cycle declined during successive cycles due to diluate and brine
392 effluent streams mixing.



395 **Figure 4:** (A) Photograph of the recirculating FDI system incorporating inlet valves (IVs)
 396 and outlet valves (OVs) that direct fluid to/from electrodes from/to a brine reservoir (BR)
 397 and a diluate reservoir (DR). (B) Top view of the cell showing valve locations. Detailed
 398 views of (C) the inlet region of the cell showing PBA symmetric electrodes sandwiching
 399 an anion exchange membrane (AEM) and (D) of valve switching actions. The time
 400 sequence of cell voltage and salt concentration used to conduct one complete cycle is
 401 shown using inset diagrams together with the timing of valve and current switching events
 402 controlled automatically. PC, OC, and NC respectively denote positive current, open
 403 circuit, and negative current.

405 Using this apparatus with the electrical/fluidic cycling sequences already
406 described, we explored the effect of PBA electrode formulation on FDI performance. As
407 a point of reference, use of this recirculating FDI apparatus with electrodes having the
408 same formulation, porosity (85%), and areal loading (5.3 mg-PBA/cm²) as in our previous
409 work (Reale et al., 2019) yielded 65% salt removal from 100 mM NaCl solution, in contrast
410 with <27% salt removal in our previous single-pass experiments. We then tested
411 electrodes cast at 250 μ m thick, solidified with WPI using room-temperature DI water, and
412 calendered to 150 μ m thick, yielding 8.7 mg-PBA/cm² and 75% porosity. The
413 corresponding experiments produced 80% salt removal, owing to the increased capacity
414 and reduced mixing volume between brine and diluate effluent from denser electrodes.
415 The third pair of electrodes was first cast and solidified by WPI in 85°C water with pH 12,
416 resulting in a 320 μ m electrode that was subsequently calendered down to 150 μ m,
417 yielding 11.1 mg-PBA/cm² and 67% porosity. This further increase of areal loading and
418 decrease of porosity produced a maximum salt removal of 90% from 100 mM influent,
419 corresponding to a productivity of 3.3 L/hour-m². A final test using 200 mM influent with
420 the 11.1 mg-PBA/cm² electrodes was performed. Because the electrodes removed more
421 salt per half-cycle, 104 mM was removed from the 200 mM influent before becoming
422 limited by mixing, due to increased ionic conductivity and decreased concentration
423 polarization improving capacity utilization.



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Figure 5: SEC versus salt removed from the dilute reservoir in units of (A) kJ/mol and (B) kWh/m³. We note that the thicker electrodes are, the greater the concentration change after each cycle due to the extended desalination time and fewer mixing opportunities during switching. (C) Thermodynamic energy efficiency for the three electrode pairs.

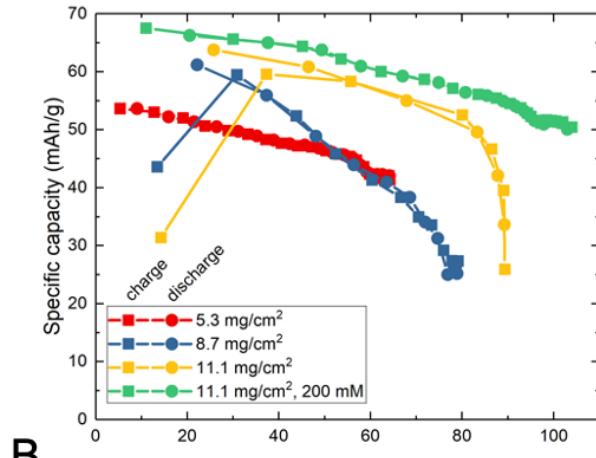
430 These high salt removals, achieved using different electrode densities, yielded
 431 different levels of specific energy consumption (SEC) to reduce influent concentration c_i
 432 to diluate concentration c_d , as shown in Fig. 5A and calculated by the equation $SEC =$
 433 $E_{total}/(c_i - c_d)V_{reservoir}$. Units of kWh/m³ are an alternate method of expressing SEC
 434 shown in Fig. 5B, omitting concentration and calculated as $SEC = E_{total}/V_{reservoir}$. We
 435 note that these SEC values are calculated based on the cumulative energy invested and
 436 the cumulative salt removal achieved, rather the energy invested and salt removed per
 437 cycle. Due to the high electronic conductivity and active particle loading of electrodes
 438 produced by the modified WPI process, the FDI system achieved excellent salt removal
 439 at an SEC that approaches the thermodynamic minimum separation energy,
 440 $SEC_{minimum} = 2RT \left(c_d \ln(c_d) + \left(\frac{1}{\gamma} - 1 \right) c_b \ln(c_b) - \frac{c_i}{\gamma} \ln(c_i) \right) / (c_i - c_d)$ where γ is the
 441 water recovery rate of 0.5 and c_b is brine concentration. As we show later, decreased
 442 energy consumption with increased electrode density is supported by its low ohmic
 443 resistance (see Fig. 7A and discussion thereof). Our previous work using the same cell
 444 operating at the same current density (1 mA/cm²) in single-pass mode showed a
 445 simulated SEC of 2.7 kJ/mol (Reale et al., 2019), while the 5.3 mg/cm² electrodes sharing
 446 the same composition have an initial SEC of 3.04 kJ/mol. As the present system reached
 447 the limit of salt removal due to the mixing of diluate and brine, further cycling consumed
 448 significantly more energy, causing SEC to rise steeply while salt removal stagnated.
 449 Because all experiments used the same flow rate and current density, the average salt
 450 absorption rate (ASAR) was initially 0.55-0.6 $\mu\text{mol}/\text{cm}^2\text{-min}$ for all three experiments.
 451 ASAR declines with the mixing effect decreasing removal each cycle, becoming 0.51
 452 $\mu\text{mol}/\text{cm}^2\text{-min}$ on average when the experiment with 11.1 mg/cm² electrodes concluded.

453 At the highest salt removal with 100 mM influent, the thermodynamic energy
454 efficiency (TEE) of the recirculating cell, defined as $TEE = SEC_{minimum}/SEC$, was 50%,
455 but TEE reached as high as 80% for the densest electrodes before mixing outweighed
456 the electrodes' ability to remove salt in a single cycle (Fig. 5C). One reason for the high
457 TEE is the high salt removal shown in Fig. 5A, where SEC for all three electrodes
458 increases linearly with salt removal during early cycles. At low salt removal, the
459 thermodynamic minimum is low compared to the experimental SECs, but as both
460 experimental and theoretical SECs increase, the ratio between them increases as well.
461 The present 100 mM influent is more ionically conductive than the 20-50 mM influents
462 used in other studies, and the denser, highly conductive electrodes created here consume
463 less energy due to smaller ohmic losses that we show subsequently. It should also be
464 noted that the cell operated at only 1 mA/cm², and SEC for similar electrodes increased
465 approximately linearly with current density in our past work (Reale et al., 2019).

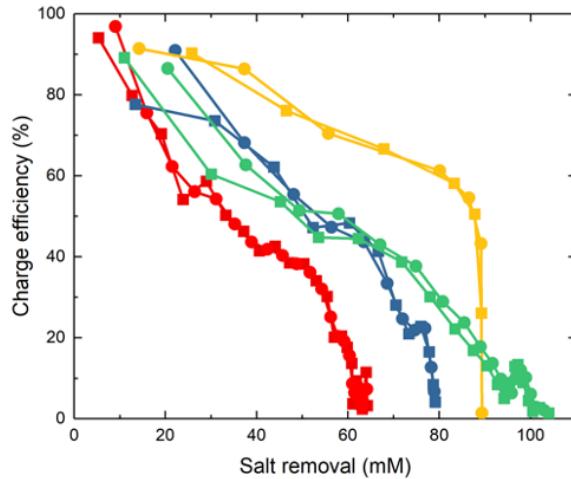
466 While such results demonstrate the practical benefits of high areal loading for
467 increased desalination performance, inspection of specific capacity attained from cycle to
468 cycle revealed trends suggesting further room for improvement. Fig. 6A shows that
469 utilization of charge capacity declined with increasing salt removal, though high areal
470 capacity electrodes retain the most capacity. When comparing the 8.7 mg/cm² and 11.1
471 mg/cm² electrodes, the former electrodes only utilize 45% of their capacity at 80% salt
472 removal, while at the same salt removal the latter electrodes retained 87% utilization. This
473 outcome is likely a result of high areal-capacity electrodes producing greater *flow*
474 *efficiency*, defined as the ratio between time for the electrodes' porous volume to be
475 replaced and the charging time (Hawks et al., 2018). Because denser electrodes have

476 smaller pore volume, feedwater is removed fast enough to supply fresh sodium ions for
477 intercalation into PBA particles, and between the lowest and highest areal loadings, flow
478 efficiency increases by 30%. The specific capacity of electrodes during experiments with
479 200 mM remained high even when the cell reached the limits of its salt removal due to
480 abundant sodium in solution. Though the flow rate was four times higher than what our
481 single-pass experiments previously used (Reale et al., 2019) (0.25 mL/min), decreasing
482 charge capacity at high salt removal suggests that gradients of salt concentration and
483 PBA state-of-charge decrease utilization. However, the disparate concentrations of salt
484 in diluate and brine streams near the end of FDI processes are also expected to produce
485 concentration polarization and ohmic polarization limiting capacity when performing
486 galvanostatic cycling within a finite cell voltage window. The inability of the system to
487 continue removing salt also causes charge efficiency to decline (Fig. 6B), eventually
488 reaching zero. During a single half-cycle, charge efficiency remains at 80-95% as salt is
489 continuously removed, but cumulative charge efficiency approaches zero due to mixing
490 counteracting salt removal.

A



B

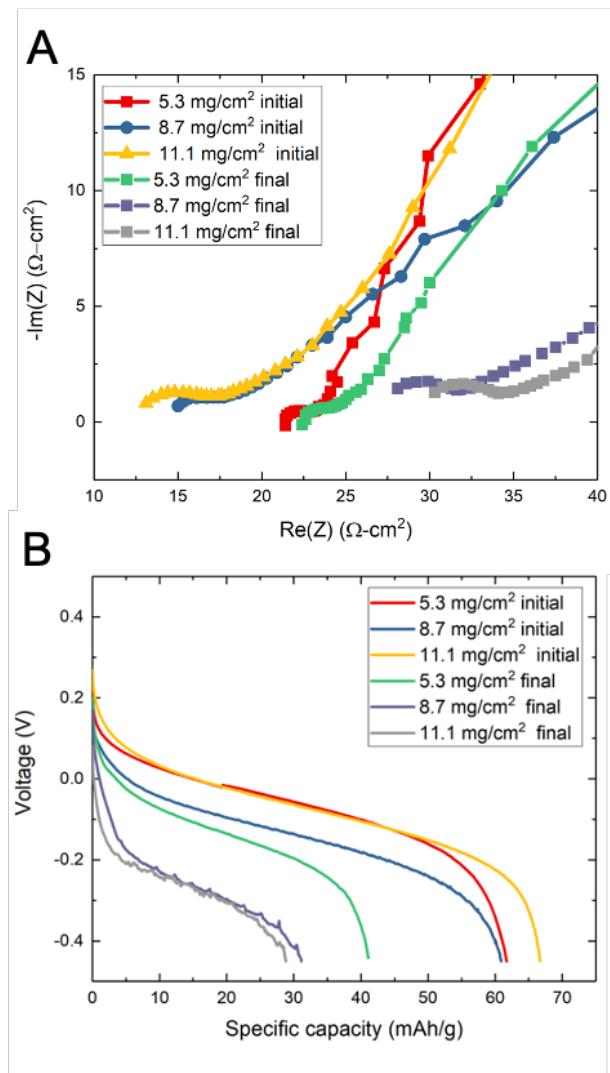


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Figure 6: (A) Specific capacity during charging and discharging versus the salt removed from the dilute stream. Initial data points for charging appear low because the electrodes were already partially charged before the experiment began. (B) Charge efficiency versus salt removed from the dilute stream for the same experiments.

497 To analyze the influence of cell impedance on desalination performance we
498 conducted electrochemical impedance spectroscopy (EIS) during the open circuit (OC)
499 periods of each cycle (Fig. 7A). These spectra show that real impedance at high
500 frequency gradually increases from cycle-to-cycle, as a result of one side of the cell
501 becoming less ionically conductive during salt removal. This effect is evidenced by the

502 voltage versus capacity curves measured during PC and NC cycling (Fig. 7B), where
 503 increased polarization pushes average cell voltage nearer to the edges of the voltage
 504 window within which cycling occurs. From the initial cycle to the final cycle, the degree
 505 of polarization increase differs with each type of electrode and the final salt removal
 506 attained. However, polarization is largest during cycles with highest salt removal, and
 507 when real impedance at high frequency is largest.



508
 509 **Figure 7:** (A) EIS curves for the three pairs of electrodes following the initial and final half-
 510 cycles of each experiment. We note that the more densely calendered electrodes show
 511 lower real impedance at high frequencies. (B) Voltage versus specific capacity curves for
 512 the initial discharging and the final charging half-cycles of the three electrode pairs,
 513 showing diminishing utilization and increasing polarization at high salt removal. For ease

514 of viewing the polarization differences, the final charging curves were made to have a
515 negative sign.

516

517 In Fig. 7A electrodes with 5.3 mg/cm² show little change in real impedance
518 between the first and final cycles, despite attaining 65% salt removal. However, real
519 impedance at high-frequency in the denser electrodes increased more than two-fold. This
520 limited change of impedance for electrodes with lower areal capacity can be understood
521 by assuming that ionic conductivity is directly proportional to salt concentration. Equation
522 1 shows the theoretical ratio between ionic resistance at a certain concentration
523 difference $R_{ion}(\Delta c)$ and the initial resistance when both anode and cathode contain
524 influent of the same concentration, neglecting the intervening AEM.

525

$$\frac{R_{ion}(\Delta c)}{R_{ion}(t = 0)} = \frac{1}{2} \left(\frac{1}{1 - \frac{\Delta c}{c_0}} + \frac{1}{1 + \frac{\Delta c}{c_0}} \right) \quad (1)$$

526 This simple relation reveals that only at 71 mM of salt removal from 100 mM does the
527 theoretical ionic resistance double. Furthermore, ionic resistance within electrodes is only
528 one source of impedance, including electronic resistance, Warburg impedance, AEM
529 impedance, and contact resistance in external circuitry. The minor influence of
530 concentration is further demonstrated in Section S2 of the SI. In any case, Fig. 7B shows
531 the increasing impedance with increasing salt removal in cell voltage curves. Average
532 voltage during the first and final half-cycles increased from 0.07V to 0.29V as the 11.1
533 mg-PBA/cm² achieved 90% salt removal, lowering utilization and limiting maximum salt
534 removal. Calendering electrodes increased electronic conductivity and reduced ohmic
535 energy losses, keeping energy consumption low at high salt removal.

536 Finally, Table 1 compares energy consumption achieved recently in FDI using
 537 symmetric electrode architectures. As each work tested an FDI system at different current
 538 densities and flow rates, the values shown are for the highest salt removal achieved. The
 539 present results show slightly higher ASAR than most cases and achieved an excellent
 540 90% salt removal, a higher fraction than any other listed. The final SEC of 10.9 kJ/mol is
 541 also high due to the mixing effect. However, at 80% salt removal an SEC of 5.7 kJ/mol
 542 was achieved, approaching the thermodynamic minimum energy of 4.5 kJ/mol for that
 543 salt removal level and reaching a record level of thermodynamic energy efficiency of 80%.

544 **Table 1:** Comparison of performance among symmetric FDI systems, including SEC,
 545 ASAR, and the percent salt removed from the initial concentration of each study. MHCF
 546 compounds denote different Prussian blue analogue materials using a certain inactive
 547 transition metal M in its lattice, while AEM and CEM denote anion and cation exchange
 548 membranes, respectively. Values in parentheses indicate those obtained at optimal
 549 thermodynamic energy efficiency (TEE), whereas the remaining values are shown for
 550 those obtained at maximum salt removal.

SEC (kJ/mol)	ASAR ($\mu\text{mol}/\text{cm}^2$ -min)	Influent salinity (mM)	Diluate salinity (mM)	% Salt Removal	Active material and separator	Other comments	Ref. no.
10.9 (5.7)	0.51	100	10 (20)	90 (80)	NiHCF/AEM /NiHCF	parallel flow- through; recirculating	this work
7.16	0.46	200	96	52	NiHCF/AEM /NiHCF	parallel flow- through; recirculating	this work
18.0	2.73	100	73	27	NiHCF/AEM /NiHCF	parallel flow-through	(Rea le et al., 2019)
10.1	0.45	20	17.5	12.5	carbon/CEM /carbon	transverse flow- through; carbon cloth modified with silane groups	(Arul rajan et al., 2019)

6.44	0.31	477	286.2	40	NiHCF/AEM /FeHCF	no flow	(Lee et al., 2017)
2.4	0.34	25	17.5	30	CuHCF/AEM/ CuHCF	flow-by	(Kim et al., 2017)
5.6	0.25	500	400	20	NiHCF/CEM/ AEM /CEM/FeHCF	Nafion-coated particles, flow-by divided by membranes	(Ahn et al., 2020 a)
27.0	6.2	500	100	80	Ag/CEM /AgCl	no flow	(Ahn et al., 2020 b)
13.76	0.534	50	35	30	CuHCF/AEM/ CuHCF	parallel flow-through	(Son et al., 2020)
21.1	3.85	50	41.7	16.7	BiOCl/CEM /BiO	parallel flow-through	(Liu et al., 2021)
23.04	0.104	20	15	25	NiHCF/AEM /NiHCF	recirculating	(Pot hana mka ndat hil et al., 2020)

552

553 **4.0 Conclusions**

554 In this work a recirculating flow-through Faradaic deionization apparatus was
 555 demonstrated and shown to produce 90% salt removed from one reservoir of 100 mM
 556 aqueous NaCl, transferring the salt into a second reservoir of brine with 50% water

557 recovery rate. In the same experiment, 80% salt removal was achieved with a TEE of
558 80%. It was shown that recirculating cell performance is limited by diluate and brine mixing
559 during valve-switching processes, the effects of which were minimized through the use of
560 an open-circuit period during which the inlet and outlet valves were switched at different
561 times. Such pause periods improved performance relative to preliminary tests, however
562 they could not totally eliminate mixing effects, resulting in an upper limit for salt removal
563 and a lower limit for specific energy consumption that is highest for porous cation
564 intercalation electrodes with high areal capacity and low porosity. We showed this by
565 fabricating electrodes using a novel wet phase inversion process incorporating a heated,
566 alkaline water bath for the first time to solidify electrodes with high areal capacity that
567 were subsequently calendered to achieve simultaneously low porosity. Impedance
568 spectra measured during the open-circuit periods of FDI cycling showed that the
569 electronic resistance of electrodes was reduced by their calendering, and flooded cell
570 experiments showed better utilization.

571 Despite the significant improvements demonstrated relative to prior work,
572 limitations other than electrode areal capacity make further increases in salt removal
573 unattainable for the current cell even with the present recirculating apparatus. For
574 example, reducing the ratio of internal fluid volume to the pore volume of electrodes is
575 likely to lead to further improvements in desalination performance, further decreasing the
576 effect of mixing between brine and diluate streams. However, the degree of apparent
577 mixing is likely to include the effects of diffusion and dispersion at interfaces between
578 diluate and brine streams. Pumping power remains significant, approximately 0.12
579 kWh/m³ for the densest electrodes during each pass of water. Over the complete

580 desalination process, this energy consumption is multiplied by over 50x due to the same
581 fluid being recirculated, making pumping energy exceed the electrical energy by an order
582 of magnitude. Thus, pumping power should be targeted as another area for improving
583 FDI performance. These results motivate other approaches to minimize losses that arise
584 due to such effects, including scale-up of the FDI cell. The current FDI cell possesses
585 pore volume, when incorporating the lowest porosity electrodes tested (67%), that is 70%
586 of the total volume within tubing and manifold supplying influent to the electrodes. An FDI
587 cell with increased capacity – constructed using more electrode area with similar areal
588 capacity – would increase total electrode pore volume in proportion to cell capacity, while
589 holding tube volume constant.

590

591 **Acknowledgments**

592 The US National Science Foundation (Award no. 1931659) and the Department of
593 Mechanical Science and Engineering at the University of Illinois at Urbana-Champaign
594 (UIUC) supported this research. Scanning electron microscopy (SEM) was performed at
595 the Materials Research Laboratory at UIUC, and X-ray computed tomography (CT) was
596 performed at the Beckman Institute for Advanced Science and Technology. We thank
597 Fitz Chem LLC for providing Ketjen black material.

598

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