

1 **Bonding performances of epoxy coatings reinforced by carbon nanotubes (CNTs) on mild steel**

2 **substrate with different surface roughness**

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9 **Abstract**

10 This paper investigated the bonding performances of epoxy coatings reinforced by carbon nanotubes (CNTs)
11 as additives on mild steel substrates. Pure epoxy and CNT-reinforced epoxy coatings on four different surface
12 roughness of steel substrate were tested using single lap shear (SLS) tests. The SLS experimental results
13 indicated that, on rougher substrates, the addition of a small percentage of CNTs (0.75% by weight) could
14 significantly improve the bonding performance and change the failure mode from adhesion fracture to partly
15 cohesive failure by improving the toughness of coatings and the interfacial adhesion between the coatings and
16 substrates. In addition, the contact angle tests and the surface characterizations using scanning electron
17 microscopy (SEM) analysis before and after fracture indicated that the wettability of coatings on steel substrates
18 improved significantly with the increase of surface roughness and mechanical interlocking was the main
19 reinforcing mechanism on rougher substrates.

20 **Keywords:** carbon nanotubes and nanofibers, resins, debonding, mechanical testing

21 **1. Introduction**

22 Polymeric coatings are one of the most commonly used coating types for protecting steel substrates from
23 corrosion as a protective barrier and it has been extensively applied in civil and transportation industries such
24 as underground, underwater and offshore infrastructures [1–3]. Epoxy resins, with high strength-to-weight ratio,
25 good environmental stability and ease of application, have become a favorable polymeric coating material in
26 the last few decades, especially for corrosion mitigation and prevention on pipeline and steel bridges. However,
27 the weak bonding performance of the epoxy coatings on steel substrates [4,5] limited their long-term

28 applications for effective and efficient corrosion protection in these fields since the debonding of the coatings
29 could occur easily under external stresses [6]. The epoxy coatings have relatively weak bonding performance
30 not only because they are brittle materials which yield relatively low ductility and resistance to crack
31 propagations [7], but also given the fact that the steel and epoxy resins are dissimilar materials, the interfacial
32 adhesion forces between steel and epoxy resins are mainly hydrogen and van der Waals force in the form of
33 secondary bonding, which are weaker compared to adhesion between similar materials [8,9]. The brittleness of
34 the pure epoxy resin results in the low bonding capacity of the coating due to poor mechanical properties [10],
35 while the secondary bonding forces between the pure epoxy and steel substrate lead to premature adhesive
36 failure in which the coating delaminates from the substrate before it fully deforms [11]. Therefore, to improve
37 the bonding performance of the epoxy coatings on steel substrates, research efforts are focused on either
38 reinforcing the mechanical properties of the epoxy coatings or enhancing the interfacial adhesion between the
39 coatings and the substrates, or both.

40 A variety of nanofillers have been incorporated into polymeric materials to enhance their mechanical,
41 electrical, and thermal properties as polymer reinforcement [12–14]. Among those nanofillers, carbon
42 nanotubes (CNTs) with extraordinarily high tensile strength and young's modulus, are expected to be promising
43 additives for epoxy coatings [15–17]. Research findings showed that adding a small percentage of CNTs into
44 the epoxy coatings as additives was a potential way to improve the bonding performance of the epoxy coatings,
45 since the CNTs reduced crack propagation within the coatings by increasing the fracture toughness [18,19]. The
46 addition of CNTs in epoxy coatings could change the failure mode from brittle adhesion failure to more
47 favorable cohesive failure [20,21]. However, the existing researches yielded inconsistent conclusions about the
48 improvement on bonding strength, even with similar weight fractions of CNTs. Some studies showed that the
49 bonding strength of 0.25% CNT-reinforced epoxy coatings was as much as 27% higher than that of pure epoxy
50 coatings [22,23], while it was reported by other studies that the addition of CNTs did not seem to significantly
51 affect the bonding strength, only a slight increase in bonding strength was obtained by CNT-reinforced epoxy
52 adhesive with the same percentage [19, 24]. Thus, it is necessary to investigate why these research findings are
53 inconsistent for the impacts of CNTs on the bonding performances of CNT-reinforced epoxy adhesive.

54 Specially, since the bonding strength of the epoxy coatings depends on both the properties of coating materials
55 and the interfacial adhesion, the inconsistency in research findings may be induced by one of these factors.

56 Most of the previous researches focused on the influences of mechanical properties on the bonding
57 performances of the CNT-reinforced epoxy and concluded that the bonding strength improvement was mainly
58 due to the improvement of mechanical properties of the CNT-reinforced epoxy, such as the reduction of crack
59 propagation and the increase of fracture toughness. However, when the interfacial adhesion is not strong enough,
60 only a small part of the coatings contributes to the bond leading to the premature adhesive failure. Thus, only
61 improving the mechanical properties of the coatings might be insufficient to achieve a firm bond if a reliable
62 interfacial adhesion does not exist between the coatings and the substrates [25, 26]. However, the impact of
63 these factors on the bonding performances of the CNT-reinforced epoxy has not been systematically
64 investigated yet.

65 The surface roughness of the steel substrate is generally recognized as the most crucial parameter
66 affecting the interfacial adhesion between the steel and the epoxy coatings. Several surface treatment methods
67 have been applied to modify the surface roughness, such as mechanical blasting [27], chemical etching [28] and
68 photolithography [29]. An adequate surface roughness is required for a good interfacial adhesion since
69 increasing the surface roughness enlarges the contact area and introduces mechanical interlocking between the
70 coatings and the substrates by strengthening the adsorption force at the interface [30,31]. Previous researches
71 only compare the bonding performances of CNT-reinforced epoxy coatings with or without a certain surface
72 treatment method [29,32,33]. It still lacks investigations on the bonding performance with a wide roughness
73 range using the same surface treatment method, since different roughness made from different treatment
74 methods might also contribute to the inconsistent results mentioned previously. Moreover, literatures only
75 focused on the bonding strength to evaluate the bonding performances of the epoxy coatings, very few studies
76 have discussed the fracture strain, which is also an important parameter for the bonding performance of the
77 epoxy coating reflecting the coating deformability.

78 To address the limitations and inconsistency of previous researches on the bonding performances of the
79 CNT-reinforced epoxy coating, for the first time as to the authors' knowledge, this paper systematically
80 investigated the bonding performances of CNT-reinforced epoxy coatings on steel substrates with four different

81 surface roughnesses using the single lap shear (SLS) tests. In addition to the traditional bonding performance
82 using the bonding strength, for the first time, the fracture elongation of the coatings was also considered for
83 bonding performance analysis. The surface morphology of the substrate before and after fracture was
84 characterized using the scanning electron microscopy (SEM) image analysis to understand the reinforcing
85 mechanisms of CNTs, and the wettability of water, pure epoxy and CNT-reinforced epoxy droplets on four
86 different substrates was also measured by the contact angle tests to evaluate the interfacial adhesion of each
87 material on each surface roughness. For the first time to the authors' knowledge, based on the experimental
88 results, this paper clearly indicated that CNT addition could strengthen the interfacial adhesion between the
89 coating matrix and the substrate. In addition, a statistical analysis was performed on the experimental data to
90 estimate the bonding strength and ultimate strain of CNT-reinforced epoxy coatings under various different
91 surface roughnesses. Although the estimation may not be universal valid, it may provide some useful
92 information on the bonding performances of CNT-reinforced epoxy coatings with any other surface roughness
93 that was not tested in this study to cover a wide roughness range.

94 **2. Experimental Setup**

95 *2.1. Materials*

96 The steel substrate was made of low carbon A36 steel (supplied by Mid America Steel Inc), which is the
97 most common structural steel in civil and transportation applications. Both pure epoxy and CNT-reinforced
98 epoxy coatings were prepared to compare their bonding performances. The epoxy coating matrix used in this
99 study was mixed thoroughly using the bisphenol A based resin and the polyamide curing agent (provided by
100 East Coast Resin) with a mixing ratio of 1:1. For the CNT-reinforced epoxy coatings, multiwalled carbon
101 nanotubes with purity higher than 95%, diameter ranging from 50 nm to 100 nm, and length ranging from 5 μm
102 to 20 μm length (supplied by Skyspring Nanomaterials Inc.) were used as CNTs reinforcement to modify the
103 epoxy coatings. For the weight fraction of CNTs in the epoxy coatings, literatures showed that the bonding
104 strength of CNT-reinforced epoxy coatings would increase with higher CNTs weight fraction till a certain
105 percentage followed by a decrease after that [34], and 0.75% by weight to the epoxy matrix was found to be the
106 optimal CNTs percentage [33,35]. Therefore, in this paper, to fabricate the CNT-reinforced epoxy coatings,

107 0.75% CNTs by weight were added into the epoxy and the mixture was mechanically stirred for 5 min, followed
108 by ultrasonic mixing for 15 min to better disperse the CNTs into the epoxy matrix.

109 *2.2. Surface preparation*

110 Four different surface roughness levels were prepared to investigate their influences on bonding
111 performances, including the smooth, fine, medium, and coarse surface conditions. Since it was reported that
112 mechanical treatment methods can quantitatively adjust the surface roughness of the substrate [36], in this paper,
113 mechanical treatment methods using sandpaper grinding and grit blasting were applied to create the four
114 different levels of roughness on steel substrates. The cleanliness of the substrate may seriously affect the bonding
115 performances of the coating. To remove any potential contaminants on the steel substrates induced before or
116 during the surface treatment process, the steel substrates were ultrasonically cleaned in pure acetone solution
117 for 15 minutes followed by compressed air cleaning before and after any surface treatments. To create the
118 surface roughness in the smooth condition level, sandpaper grinding with 60-grit, 120-grit, 220-grit and 400-
119 grit was conducted on the substrates successively. To prepare the surface roughness in the fine, medium, and
120 coarse condition levels, the steel substrates were grit blasted with 20-grit, 36-grit and 60-grit aluminum oxides
121 with a blasting pressure of 500 kPa and a standoff distance of 150 mm.

122 *2.3. Roughness measurements*

123 Right after all the surface treatments, the surface roughness of the steel substrates was measured using an
124 PCE-RT 1200 roughness tester (supplied by PCE Instruments) following the ASTM D7127-17 standard [37].
125 Three roughness parameters were collected to evaluate the surface roughness, including the average roughness,
126 R_z , which is the arithmetic mean of the largest individual seam depths, the arithmetic mean roughness, R_a , which
127 is the arithmetic mean of the absolute values of the profile deviations within the reference line, and the
128 maximum roughness, R_t , which is the distance between the highest and the lowest points on the surface [38].
129 On each substrate, the roughness measurements were performed on five different locations and the final
130 roughness values were determined by the average of these five measurements for a more statistically accurate
131 result. In addition, the surface morphology of the substrates with different roughness levels was also studied
132 using the SEM image analysis.

133 2.4. *Contact angle tests*

134 In addition, since the wettability of the steel substrates is an essential surface characterization for the
135 bonding performance of the epoxy-based coatings, this paper also studied the wettability of the steel substrates
136 with different surface roughness using the contact angle tests. In total, twelve different combinations of contact
137 angle tests from four different surface roughness levels (smooth, fine, medium, and coarse conditions) and three
138 different liquid materials (water, pure epoxy, and CNT-reinforced epoxy) were performed. Water droplets were
139 used to estimate the surface energy which is a substantial property of different substrates, while pure epoxy
140 resin and CNT-reinforced epoxy resin were used to evaluate their wettability on all the four different substrates.
141 The contact angle tests were carried out by the FTA1000 Drop Shape Instrument B Frame Analyzer System
142 (supplied by First Ten Angstroms, Inc.) following the ASTM D7334-08 standard [39]. To be statistically valid,
143 three drops of each liquid were placed on each substrate and two angle measurements were made on each edge
144 of the droplets within 30 seconds after depositing the droplet.

145 2.5. *SLS tests*

146 The bonding performances of epoxy-based coatings was studied by SLS tests. The SLS test specimens
147 were designed according to the ASTM D1002-10 standard [40] as shown in Figure 1. Two mild steel sheets
148 with a length of 101.6 mm, a width of 25.4 mm, and a thickness of 3.18 mm were bonded together by pure
149 epoxy or CNT-reinforced epoxy coatings in the overlap area of 12.7 mm in length. To ensure the tensile loading
150 direction coincided with the central line of the coating and the pure shear stress was the dominant stress state
151 within the coating, on the other ends of the two steel sheets, steel attachments with the same thickness as the
152 steel sheets and a length of 25.4 mm were bonded using the same coating materials and thickness as in the
153 overlap area. The surface treatments for different roughness were only performed on the overlap area of each
154 specimen. A previous study [41] found that the optimum epoxy thickness for epoxy coating on steel surfaces
155 was in between 0.4 and 0.5 mm because thicker coatings may lead to weaker bonding strength and thinner
156 coatings were prone to have excessive strength data deviation. Thus, the coating thickness in this study was
157 controlled to be around 0.5 mm by using steel shims. The two epoxy-based coating materials were applied on
158 both surfaces of the steel sheets and assembled within 24 hours after surface treatments, followed by a 24-hour
159 curing at 32 °C and a curing for 7 days at room temperature before the SLS tests.

160 Table 1 shows the test matrix for SLS tests. Eight testing groups were prepared, including two different
161 coating materials (both pure epoxy and CNT-reinforced epoxy coatings) and four different surface roughness
162 (smooth, fine, medium, and coarse levels). For each testing group, five valid specimens were made, resulting
163 in a total of 60 specimens. All the SLS specimens were tested using the MTS Flex Test® SE loading frame as
164 shown in Figure 2 under monotonic tensile loading till failure. The SLS tests were conducted by the
165 displacement control mode at a loading rate of 1.3 mm/min. The real-time tensile load and displacement of each
166 specimen as well as corresponding load-displacement curves were recorded. It should be noted that the tensile
167 displacement of the specimen was equal to the elongation of the epoxy-based coatings. To calculate the bonding
168 strength and ultimate strain of the epoxy-based coatings, the recorded tensile load and displacement were
169 converted into shear stress and strain, so the load-displacement curves could be transformed into the stress-
170 strain curves. For the convenience of comparison among different testing groups, the stress-strain curves of five
171 individual specimens in one testing group were fitted mathematically into one average representative curve
172 using the Trace Interpolation algorithm. The shape of the fitted curve was precisely similar to the five
173 experimental curves and each point on the fitted curve was within a certain tolerance from the experimental
174 curves. Moreover, the SEM images were also taken on the fracture surfaces of SLS specimens after testing.

175 3. Experimental Results

176 3.1. Surface characterizations

177 Table 2 shows the results of the average roughness (R_z , R_a and R_t) as well as its standard deviations (STDs)
178 for the four different surface roughness levels. The STDs in percentage were calculated as the fraction of STDs
179 by the average values. As expected, the surface roughness increased remarkably with higher roughness levels
180 which were resulted from smaller girt size, and the same trend was seen in all the three parameters. Since the
181 R_z and R_t only consider the characteristics of the peaks and valleys on the substrate, and the R_a gives a more
182 comprehensive description of the profile including the height of every single point on the substrate as a
183 commonly used international roughness parameter. Although all the three surface roughness parameters were
184 recorded, in following discussions and analysis of this paper, only R_a was used to represent the surface
185 roughness. Thus, the average surface roughness of the four different roughness levels of smooth, fine, medium,
186 and coarse conditions achieved in this study, were 0.231 μm , 3.528 μm , 5.272 μm , and 8.457 μm , respectively.

187 The surface characterizations of the steel substrates can also be observed visually by surface morphology
188 from the SEM analysis. Figures 3(a ~ d) show the SEM images of the four roughness levels at a magnification
189 of 100X. The ground surface at the smooth roughness level was observed to be remarkably flat with minor
190 interfacial scratches. The surface of the substrate at the fine level of roughness was densely filled with small
191 bumps and holes, while at the medium roughness level the surface was majorly full of small irregularities except
192 a few higher hills and deeper valleys and the coarse surface was clearly filled with more clearly visible hills
193 and valleys. Higher bumps and deeper holes magnified the height difference of the profile leading to higher
194 value of R_s and R_a , and denser distribution of these irregularities shorten the distance between peaks and bottoms
195 resulting in smaller R_t , which verified the results of quantitative roughness measurement in Table 2.

196 For the wettability test results from the contact angle tests, Figures 4(a ~ d) show the typical appearances
197 and contact angles of water droplets on four substrates with different roughness. Although all the four contact
198 angles were less than 90°, indicating that all four substrates belonged to hydrophilic surfaces, the variation of
199 the contact angles with four surface roughness levels shared the same trend as the roughness values increased.
200 The contact angle of the smooth substrate was 74.72° which was apparently larger than those on the other
201 substrates. Contact angles were noticed to be reduced from fine to medium and coarse substrates, of 59.84°,
202 55.67° and 48.11°, respectively. A smaller contact angle of water droplets with higher surface roughness
203 indicated a higher wettability. Since the rougher substrates could have higher surface energy and more contact
204 areas for the coating materials to contact with, epoxy-based coatings on rougher substrates were expected to
205 have better bonding performance than on smoother substrates.

206 3.2. SLS test results

207 Figures 5(a, b) plot all the average stress-strain curves of pure and CNT-reinforced epoxy coatings with
208 regard to the four different surface roughness. Figure 5(a) shows that the stress and strain of pure epoxy coatings
209 exhibited evident linear relationship until a sudden failure occurred at the peak stress, regardless of the surface
210 roughness levels. As for the CNT-reinforced epoxy coatings, similar to those of pure epoxy coatings, the stress-
211 strain curves went up approximately linearly before peak stress. But after the peak stress, only the curves of
212 smooth surfaces followed the same pattern as the pure epoxy coatings ended with a sudden failure. The curves
213 of the rest three rougher surfaces dropped gradually from the peak stress to the failure, indicating an obvious

214 nonlinear behavior. The three curves of CNT-reinforced epoxy coatings on the rougher surfaces showed typical
215 stress-strain relation as for ductile materials with strain continually growing at a relatively stable stress level
216 rather than a sharp failure for brittle materials. The nonlinearity illustrated in the curves was an indication of
217 the plastic behavior of the epoxy coating with the addition of CNTs on the rougher steel substrates.

218 The average bonding strength and fracture elongation were compared in details as shown in Table 3 for all
219 eight testing groups. The bonding strengths of the coatings were determined as the peak shear stresses from the
220 stress-strain curves and the fracture elongations of the coatings were reflected by the ultimate strains, which
221 was identified as the strains when the curve experienced a rapid stress drop. As shown in Table 3, almost all
222 the STDs in one testing group were smaller than 10% for both bonding strength and ultimate strains, suggesting
223 the consistency of the SLS tests.

224 **4. Data analysis and discussion**

225 *4.1 Influences of the addition of CNTs in epoxy coatings*

226 Figures 6(a, b) illustrate the bar chart comparison of bonding strength and ultimate strain between the pure
227 and the CNT-reinforced epoxy coatings. The addition of CNTs (0.75%) showed greatly improvements of
228 boning performances in both the bonding strength and ultimate strain, but the improvements varied with
229 different surface roughness levels. On the smooth substrate, compared to pure epoxy, the addition of CNTs
230 increased the bonding strength and ultimate strain by around 56% and 84%, while on the fine surfaces, while
231 the enhancements of bonding strength and ultimate strain by adding CNTs were much more significant by
232 around 123% and 382%, respectively. On the medium and coarse substrates, the enhancement of bonding
233 strength by CNTs were around 70% compared to pure epoxy, but the CNTs reinforcement in epoxy improved
234 the ultimate strain significantly by around 280%. It is also worth mentioning that the increments in ultimate
235 strain were much more pronounced than those in bonding strength, which was largely due to the plastic
236 deformation created by CNT-reinforced epoxy coatings.

237 Figure 6(c) demonstrates the toughness of the pure and the CNT-reinforced epoxy coatings with the four
238 different surface treatments. The toughness is defined as the ability of deforming plastically and absorbing
239 energy before fracture, which could be evaluated by the area under the stress-strain curve as in Figure 3. From
240 Figure 6(c), it can be seen that the CNTs in epoxy only increased the toughness slightly on the smooth substrate,

241 but greatly on the other three rougher substrates. As a result of higher toughness, the CNT-reinforced epoxy
242 coatings generated more plastic deformation resulting in higher increases in ultimate strain and smaller
243 increases in bonding strength, compared to pure epoxy.

244 Figures 7(a, b) illustrate the typical fracture surfaces of SLS specimens with or without CNTs on the smooth
245 substrates. No obvious differences were observed on the fracture surfaces between the pure and the CNT-
246 reinforced coatings on the smooth substrates. Both coatings had fractures occurred at the coating-substrate
247 interfaces with all the coatings attached on the bottom surfaces of the specimens and no visible epoxy remaining
248 on the other side, indicating a typical adhesive failure. For an adhesive failure, the interfacial adhesion instead
249 of the coating mechanical properties played the dominating role for the bonding performance. Only the coatings
250 near the coating-substrate interface contributed to the bond, while the rest large part of the coatings did not
251 contribute a lot to the bond before the catastrophic failure occurred on the interface. Therefore, on the smooth
252 substrates, increasing the mechanical properties of the epoxy-based coatings (such as toughness) by adding
253 CNTs had little influences on the failure mode. Although the failure modes remained the same for both the pure
254 epoxy and CNT-reinforced epoxy on smooth substrates as indicated in Figure 7, the bonding strengths and
255 ultimate strains of the CNT-reinforced epoxy coatings on smooth surfaces were still moderately improved
256 compared to the pure epoxy coatings as shown in Figure 6(a, b). The resulted increases on the bonding
257 performance might be benefited from the improvement of the interfacial adhesion between the coatings and the
258 substrates with the addition of the CNTs.

259 To study the CNTs contributions to the interfacial adhesion between the epoxy-based coatings and steel
260 substrates, Figures 8(a ~ h) display the contact angles of the pure and the CNT-reinforced epoxy on the steel
261 substrates with the four different surface roughness levels. The contact angles on smooth substrates decreased
262 from 53.42° for pure epoxy coatings to 52.13 ° for CNT-reinforced epoxy coatings, and similar reductions in
263 contact angles were also observed in the other three rougher substrates, indicating a higher wettability by the
264 CNTs reinforcement. Higher wettability by the addition of CNTs could increase the contact area between the
265 CNT-reinforced epoxy coating and the substrate. In addition, the epoxy which flowed into the irregularities of
266 the substrate was also reinforced by the CNTs, which enhanced the connection of the epoxy and the substrate
267 as well. The reductions of the contact angle for the CNTs reinforcement in epoxy coatings confirmed with the

268 findings from failure mode analysis in Figures 7(a, b) that the CNTs could improve the interfacial adhesion
269 between the coatings and substrates, resulting in the improvements in the bonding performances as shown in
270 Figures 6(a, b).

271 Figures 9(a, c) compare the overlap areas of the pure and the CNT-reinforced coatings on coarse substrates
272 after SLS tests. For the pure epoxy coatings, similar as the fracture surfaces on the smooth surfaces, there was
273 no sign of epoxy left on the top surface and all the epoxy coating on the bottom surface was free of any
274 noticeable scars or cracks as shown in Figure 9(a), indicating that the adhesive failure was the dominant failure
275 mode on the coarse substrates as well. However, for the CNT-reinforced epoxy coatings, as shown in Figure
276 9(c), a main crack was found in the middle of the coating and the coating material was left on both top and
277 bottom surfaces. As fracture occurred within the coating layer in the cracking area, partly cohesive failure was
278 achieved on the coarse-blasted substrates with the addition of CNTs. Figures 9(b, d) further compare the SEM
279 images of bottom surfaces in the overlap areas with the pure and the CNT-reinforced coatings on coarse
280 substrates after testing. The fracture surface of the CNT-reinforced coatings was observed to be much rougher
281 than that of the pure epoxy coatings, indicating a sign of more plastic deformations and consequently more
282 fracture energy consumption with the addition of CNTs.

283 To further investigate the reinforcing mechanism of CNTs in epoxy coatings, the SEM analysis under
284 higher magnifications was conducted on the fracture surface and the main crack in Figure 9(b) of the CNT-
285 reinforced epoxy coatings, as shown in Figures 10(a ~ d). The pulling-out of CNTs as shown in Figures 10(a)
286 and (b) was noticed as an important reinforcing mechanism, which could improve the bonding performance of
287 the epoxy-based coatings. It required considerable energy to pull out the CNTs from the surrounding epoxy
288 coatings, leading to the higher toughness of the CNT-reinforced epoxy coatings than the pure epoxy coatings.
289 Figures 10(c) and (d) illustrate the CNT clusters on the main crack. Even though the weight fraction of CNTs
290 was optimized and the ultrasonic mixing was used in mixing the CNTs in the epoxy matrix, the CNTs were still
291 noticed to be not uniformly dispersed in the epoxy matrix with CNTs agglomerated into clusters as shown in
292 Figure 10(c). The aggregation of CNTs was generated primly due to the high viscosity of epoxy and high surface
293 energy of CNTs [18], which had a detrimental effect on the bonding performance. According to the literature
294 [42], CNTs can be divided into three levels based on the unit structure, namely individual CNTs, CNT bundles

295 (close-packed CNTs) and CNT fibers (an assembly of CNT bundles). The strength and toughness of the CNT
296 clusters reduced significantly, as the aggregation of CNTs become larger. These CNTs clusters consisted of
297 both CNT bundles and CNT fibers, which reduce the reinforcing mechanism of CNTs and also caused the local
298 stress concentration. The adverse effect of CNT clusters as defects or imperfections led to the rapid growth of
299 the main crack, which eventually restricted the improvement of CNTs on the bonding performances. On ideal
300 condition when the CNTs are uniformly dispersed in the epoxy matrix, CNT aggregation would not produce
301 any defects and initial voids would be all filled by CNTs. Thus, it is expected that the bonding performance of
302 CNT-reinforced epoxy coating would get tremendously further improved if all the imperfections were
303 eliminated within the coating layer. Thus, how to control the CNT aggregation and improve the dispersion into
304 epoxy resin is a prospective issue, while more in-depth investigations on this aspect are beyond the scope of
305 this study. Future efforts are needed to improve the dispersion of CNTs in the epoxy matrix to reduce the
306 imperfections of the resulted coatings.

307 *4.2 Influences of surface roughness for the CNT-reinforced epoxy coatings*

308 Figure 8 showed that the increase of the surface roughness could reduce the contact angles of the CNT-
309 reinforced epoxy, thus, improve the wettability of the substrates, resulting in improvements in the interfacial
310 adhesion to benefit the overall bonding performances. To further investigate the influences of surface roughness
311 on the bonding performance of the CNT-reinforced epoxy coatings, Figure 11 plots the changes of the bonding
312 strengths and ultimate strains of the CNT-reinforced epoxy coatings with the changes of the surface roughness
313 parameter, R_a . In general, the changing trends of the bonding strengths and ultimate strains approximately
314 followed a logarithmic pattern as the increase of surface roughness, with both of the curves rising rapidly at
315 lower surface roughness and then growing at a much slower rate at larger surface roughness. On the smooth
316 substrates with the surface roughness of $0.231 \mu\text{m}$, the bonding strengths and ultimate strains were greatly lower
317 than the other three rougher substrates owing to the lack of interfacial adhesion. Insufficient interfacial adhesion
318 might result in the moderate improvements in the literature. As the surface roughness increased from $0.231 \mu\text{m}$
319 on smooth substrates to $3.528 \mu\text{m}$ on the fine substrates, significant increases in bonding strength and ultimate
320 strain were noted due to the improvement of the interfacial adhesion. Stronger interfacial adhesion could prevent
321 premature adhesive failure and allow the coatings to deform plastically as demonstrated in Figures 5 and 9.

322 When the steel substrates were further roughened from 3.528 μm on the fine substrates to 5.272 μm on the
323 medium substrates and 8.457 μm on the coarse substrates, the bonding strengths barely changed, but the ultimate
324 strains increased about 20% and 11%, from fine to coarse substrates, respectively. A much less significant
325 changes were observed for both the bonding strengths and ultimate strains, compared to changing the surface
326 roughness from smooth to fine conditions, indicating that although the surface roughness was of vital
327 importance to the bonding performances of the CNT-reinforced epoxy coatings, the influence was more crucial
328 at lower roughness levels. When surface roughness was sufficient to provide a good interfacial adhesion, further
329 increasing the surface roughness become much less effective.

330 To enable an estimation of bonding performances under all different surface roughness other than the
331 values tested in this study, various fitting approaches were performed based on the obtained data in Figure 11
332 and Table 3. Although there were some differences between the changes of the bonding strengths and ultimate
333 strains for the CNT-reinforced epoxy coatings, the best fitted curves of both bonding strength and ultimate
334 strain could be expressed into a logarithmic equation as below:

$$y = a/\{1 + \exp[-k(x - x_c)]\} \quad (1)$$

335 in which, a is the curve's maximum value, k is the logistic growth rate or steepness of the curve, and x_c is the
336 value of the sigmoid's midpoint. The fitted curves using Eq. (1) was also included in Figure 11. In addition,
337 Table 6 shows the detailed fitted parameters in Eq. (1) as well as the adjusted R-squared to evaluate the goodness
338 of the fittings. With all the R^2 being precisely close to 1 as shown in Table 6, Figure 11 also shows that all the
339 traces of two fitting curves staying within the STDs of measured data points on original curves, indicating an
340 effective fitting for the experimental data for future prediction use.

341 In addition, Figures 12(a, b) further compare the SEM images of the top fracture surfaces for the CNT-
342 reinforced epoxy coatings on smooth and coarse substrates. On the smooth substrates, there was no epoxy on
343 the top surface and the fracture substrate after the SLS test was very similar to the substrate before applying the
344 coatings. However, on the coarse substrates, it was evident that some of the CNT-reinforced epoxy penetrated
345 into the irregularities of the substrates, as the indication of mechanical interlocking. The stronger interfacial
346 adhesion by mechanical interlocking might contribute to the improvements of the bonding performances of the
347 CNT-reinforced epoxy coating on rougher substrates as shown in Figure 11.

348 **5. Conclusions**

349 This paper investigated the bonding performances of epoxy coatings with and without CNT reinforcement
350 on mild steel substrates fabricated with four different surface roughness. According to the experimental results,
351 the following concluding remarks could be drawn:

352 (1) The addition of CNTs could significantly increase the bonding strengths and ultimate strains as a result
353 of great improvement in the toughness of the epoxy coatings. Higher ability of plastic deformation and pulling-
354 out of CNTs with improved fracture energy consuming efficiency might be the reinforcing mechanisms when
355 interfacial adhesion between the epoxy coatings and steel substrates was strong enough.

356 (2) When lacking sufficient interfacial adhesion due to low surface roughness, the bonding performance of
357 the CNT-reinforced epoxy coatings could still be improved because the addition of CNTs could improve the
358 interfacial adhesion as indicated by the smaller contact angles of the CNT-reinforced epoxy compared to the
359 pure epoxy, although the improvement was less significant.

360 (3) The surface roughness had a positive influence on the bonding performances of the CNT-reinforced
361 epoxy coatings by introducing mechanical interlocking to enhance the interfacial adhesion. The positive
362 influence of surface roughness was more significant on smoother substrates when lacking sufficient interfacial
363 adhesion. When surface roughness was sufficient to provide a good interfacial adhesion, further increasing the
364 surface roughness become much less effective.

365 (4) The failure mode could only be changed from adhesive failure of the pure epoxy coatings to partly
366 cohesive failure of the CNT-reinforced epoxy coatings on highly rough substrates due to the aggregation of
367 CNTs even after mechanical stirring and ultrasonic mixing.

368 Although the bonding strengths and ultimate strains of CNT-reinforced epoxy coatings on rough substrates
369 were satisfactory, this study also showed that it was very challenging to achieve a complete cohesive failure
370 only with the addition of CNTs in the epoxy because the CNTs tended to aggregate together in the epoxy
371 coatings which introduced more defects in the coatings. In the future, researches are needed to optimize the
372 dispersion of CNTs in the epoxy coatings and eventually to further increase the bonding performances.

373 **CRediT authorship contribution statement**

374 **Dawei Zhang**: Data curation, Methodology, Formal analysis, Investigation, Writing - original draft,
375 Writing - review & editing. **Ying Huang**: Project administration, Funding acquisition, Supervision, Writing -
376 review & editing. **Yechun Wang**: Data curation.

377 **Declaration of interest**

378 The authors declare that they have no known competing financial interests or personal relationships that
379 could have appeared to influence the work reported in this paper.

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383 the sponsors.

384 **Reference**

385 [1] Benea L, Mardare L, Simionescu N. Anticorrosion performances of modified polymeric coatings on
386 E32 naval steel in sea water. *Prog Org Coatings* 2018;123:120–7.
387 <https://doi.org/10.1016/j.porgcoat.2018.06.020>.

388 [2] Laurentiu M, Lidia B, Eliza D, Valentin D. Polymeric coatings used against marine corrosion of naval
389 steel EN32. *Key Eng. Mater.*, vol. 699, Trans Tech Publications Ltd; 2016, p. 71–9.
390 <https://doi.org/10.4028/www.scientific.net/KEM.699.71>.

391 [3] Lyon SB, Bingham R, Mills DJ. Advances in corrosion protection by organic coatings: What we
392 know and what we would like to know. *Prog Org Coatings* 2017;102:2–7.
393 <https://doi.org/10.1016/j.porgcoat.2016.04.030>.

394 [4] Zhang L, Wu H, Zheng Z, He H, Wei M, Huang X. Fabrication of graphene oxide/multi-walled
395 carbon nanotube/urushiol formaldehyde polymer composite coatings and evaluation of their physico-
396 mechanical properties and corrosion resistance. *Prog Org Coatings* 2019;127:131–9.
397 <https://doi.org/10.1016/j.porgcoat.2018.10.026>.

398 [5] Baltzis D, Orfanidis S, Lekatou A, Paipetis AS. Stainless steel coupled with carbon nanotube-
399 modified epoxy and carbon fibre composites: Electrochemical and mechanical study. *Plast Rubber*

400 Compos 2016;45:95–105. <https://doi.org/10.1080/14658011.2016.1144339>.

401 [6] Parhizkar N, Ramezanadeh B, Shahrabi T. Corrosion protection and adhesion properties of the epoxy
402 coating applied on the steel substrate pre-treated by a sol-gel based silane coating filled with amino
403 and isocyanate silane functionalized graphene oxide nanosheets. *Appl Surf Sci* 2018;439:45–59.
404 <https://doi.org/10.1016/j.apsusc.2017.12.240>.

405 [7] García GL, López-Ríos V, Espinosa A, Abenojar J, Velasco F, Toro A. Cavitation resistance of
406 epoxy-based multilayer coatings: Surface damage and crack growth kinetics during the incubation
407 stage. *Wear* 2014;316:124–32. <https://doi.org/10.1016/j.wear.2014.04.007>.

408 [8] Bahlakeh G, Ramezanadeh B. A Detailed Molecular Dynamics Simulation and Experimental
409 Investigation on the Interfacial Bonding Mechanism of an Epoxy Adhesive on Carbon Steel Sheets
410 Decorated with a Novel Cerium-Lanthanum Nanofilm. *ACS Appl Mater Interfaces* 2017;9:17536–51.
411 <https://doi.org/10.1021/acsami.7b00644>.

412 [9] Semoto T, Tsuji Y, Yoshizawa K. Molecular understanding of the adhesive force between a metal
413 oxide surface and an epoxy resin. *J Phys Chem C* 2011;115:11701–8.
414 <https://doi.org/10.1021/jp202785b>.

415 [10] Moulds RJ, Baldwin TR. Toughened adhesives for structural applications. *Int J Adhes Adhes* 1983;
416 vol. 3:203–207. [https://doi.org/10.1016/0143-7496\(83\)90095-7](https://doi.org/10.1016/0143-7496(83)90095-7)

417 [11] Kim KS, Yoo JS, Yi YM, Kim CG. Failure mode and strength of uni-directional composite single lap
418 bonded joints with different bonding methods. *Compos Struct* 2006;72:477–85.
419 <https://doi.org/10.1016/j.compstruct.2005.01.023>.

420 [12] Lee JH, Rhee KY, Park SJ. Silane modification of carbon nanotubes and its effects on the material
421 properties of carbon/CNT/epoxy three-phase composites. *Compos Part A Appl Sci Manuf*
422 2011;42:478–83. <https://doi.org/10.1016/j.compositesa.2011.01.004>.

423 [13] Jojibabu P, Jagannatham M, Haridoss P, Janaki Ram GD, Deshpande AP, Bakshi SR. Effect of
424 different carbon nano-fillers on rheological properties and lap shear strength of epoxy adhesive joints.
425 *Compos Part A Appl Sci Manuf* 2016;82:53–64. <https://doi.org/10.1016/j.compositesa.2015.12.003>.

426 [14] Park SM, Shon MY. Effects of multi-walled carbon nano tubes on corrosion protection of zinc rich

427 epoxy resin coating. *J Ind Eng Chem* 2015;21:1258–64. <https://doi.org/10.1016/j.jiec.2014.05.042>.

428 [15] Yu S, Tong MN, Critchlow G. Use of carbon nanotubes reinforced epoxy as adhesives to join
429 aluminum plates. *Mater Des* 2010;31. <https://doi.org/10.1016/j.matdes.2009.11.045>.

430 [16] Seo DW, Lim JK. Tensile, bending and shear strength distributions of adhesive-bonded butt joint
431 specimens. *Compos Sci Technol* 2005;65:1421–7. <https://doi.org/10.1016/j.compscitech.2004.12.013>.

432 [17] Khoramishad H, Khakzad M. Toughening epoxy adhesives with multi-walled carbon nanotubes. *J*
433 *Adhes* 2018;94:15–29. <https://doi.org/10.1080/00218464.2016.1224184>.

434 [18] Razavi SMJ, Ayatollahi MR, Nemati Giv A, Khoramishad H. Single lap joints bonded with structural
435 adhesives reinforced with a mixture of silica nanoparticles and multi walled carbon nanotubes. *Int J*
436 *Adhes Adhes* 2018;80:76–86. <https://doi.org/10.1016/j.ijadhadh.2017.10.007>.

437 [19] Konstantakopoulou M, Kotsikos G. Effect of MWCNT filled epoxy adhesives on the quality of
438 adhesively bonded joints. *Plast Rubber Compos* 2016;45:166–72.
439 <https://doi.org/10.1080/14658011.2016.1165788>.

440 [20] Cheng F, Hu Y, Lv Z, Chen G, Yuan B, Hu X, et al. Directing helical CNT into chemically-etched
441 micro-channels on aluminum substrate for strong adhesive bonding with carbon fiber composites.
442 *Compos Part A Appl Sci Manuf* 2020;135. <https://doi.org/10.1016/j.compositesa.2020.105952>.

443 [21] Wang B, Bai Y, Hu X, Lu P. Enhanced epoxy adhesion between steel plates by surface treatment and
444 CNT/short-fibre reinforcement. *Compos Sci Technol* 2016;127:149–57.
445 <https://doi.org/10.1016/j.compscitech.2016.03.008>.

446 [22] Amitkumar R, Asokan R, Jhanji KP, Das D, Sai NV. Investigation of tensile properties of
447 carbon/epoxy composite joints with and without carbon nano-tubes. *Int J Veh Struct Syst*
448 2019;11:209–13. <https://doi.org/10.4273/ijvss.11.2.19>.

449 [23] Han S, Meng Q, Araby S, Liu T, Demiral M. Mechanical and electrical properties of graphene and
450 carbon nanotube reinforced epoxy adhesives: Experimental and numerical analysis. *Compos Part A*
451 *Appl Sci Manuf* 2019;120:116–26. <https://doi.org/10.1016/j.compositesa.2019.02.027>.

452 [24] Prolongo SG, Gude MR, Ureña A. Rheological behaviour of nanoreinforced epoxy adhesives of low
453 electrical resistivity for joining carbon fiber/epoxy laminates. *J Adhes Sci Technol* 2010;24:1097–

454 112. <https://doi.org/10.1163/016942409X12584625925060>.

455 [25] Marshall SJ, Bayne SC, Baier R, Tomsia AP, Marshall GW. A review of adhesion science. *Dent*
456 *Mater* 2010;26. <https://doi.org/10.1016/j.dental.2009.11.157>.

457 [26] Zhang D, Huang Y. Influence of surface roughness and bondline thickness on the bonding
458 performance of epoxy adhesive joints on mild steel substrates. *Prog Org Coatings* 2021;153:106135.
459 <https://doi.org/10.1016/j.porgecoat.2021.106135>.

460 [27] Rajput A, Ak M, Kim SJ, Noh SH, Park JH, Paik JK. Effects of the surface preparation on the life of
461 epoxy coating in steel ship plates: an experimental study. *Ships Offshore Struct* 2019;14:199–206.
462 <https://doi.org/10.1080/17445302.2019.1565072>.

463 [28] Lei Y, Ma Y, Zhang B, Lei B, Li Y. Enhancement the adhesion between epoxy coating and rusted
464 structural steel by tannic acid treatment. *Acta Metall Sin (English Lett)* 2014;27:1105–13.
465 <https://doi.org/10.1007/s40195-014-0132-5>.

466 [29] Gude MR, Prolongo SG, Gómez-Del Río T, Ureña A. Mode-I adhesive fracture energy of carbon
467 fibre composite joints with nanoreinforced epoxy adhesives. *Int J Adhes Adhes* 2011;31:695–703.
468 <https://doi.org/10.1016/j.ijadhadh.2011.06.016>.

469 [30] Hunter R, Ibáñez N, Möller J, Betancourt R, Mora T, Diez E, et al. Influence of roughness on the
470 mechanical adhesion of single lap joints. *J. Adhes.*, vol. 88, 2012, p. 376–90.
471 <https://doi.org/10.1080/00218464.2012.660107>.

472 [31] Fernando D, Teng ; J G, Asce M, Yu ; T, Zhao XL, Asce F. Preparation and Characterization of Steel
473 Surfaces for Adhesive Bonding 2013. [https://doi.org/10.1061/\(ASCE\)CC.1943-5614](https://doi.org/10.1061/(ASCE)CC.1943-5614).

474 [32] Yun IH, Kim WS, Kim K, Jung JM, Lee JJ, Jung HT. Highly enhanced interfacial adhesion properties
475 of steel-polymer composites by dot-shaped surface patterning. *J. Appl. Phys.*, vol. 109, 2011.
476 <https://doi.org/10.1063/1.3567113>.

477 [33] Kumar A, Kumar K, Ghosh PK, Rathi A, Yadav KL, Raman. MWCNTs toward superior strength of
478 epoxy adhesive joint on mild steel adherent. *Compos Part B Eng* 2018;143:207–16.
479 <https://doi.org/10.1016/j.compositesb.2018.01.016>.

480 [34] Ayatollahi MR, Nemati Giv A, Razavi SMJ, Khoramishad H. Mechanical properties of adhesively

481 single lap-bonded joints reinforced with multi-walled carbon nanotubes and silica nanoparticles. *J*
482 *Adhes* 2017;93:896–913. <https://doi.org/10.1080/00218464.2016.1187069>.

483 [35] Sydlik SA, Lee JH, Walish JJ, Thomas EL, Swager TM. Epoxy functionalized multi-walled carbon
484 nanotubes for improved adhesives. *Carbon N Y* 2013;59:109–20.
485 <https://doi.org/10.1016/j.carbon.2013.02.061>.

486 [36] Sen D, Chavan NM, Rao DS, Sundararajan G. Influence of grit blasting on the roughness and the
487 bond strength of detonation sprayed coating. *J. Therm. Spray Technol.*, vol. 19, 2010, p. 805–15.
488 <https://doi.org/10.1007/s11666-010-9476-1>.

489 [37] ASTM. D7127-13. Standard test method for measurement of surface roughness of abrasive blast
490 cleaned metal surfaces using a portable stylus instrument. *ASTM Int* 2014:1–6.
491 <https://doi.org/10.1520/D7127-13.2>.

492 [38] Gadelmawla ES, Koura MM, Maksoud TMA, Elewa IM, Soliman HH. Roughness parameters. *J*
493 *Mater Process Technol* 2002;123:133–45. [https://doi.org/10.1016/S0924-0136\(02\)00060-2](https://doi.org/10.1016/S0924-0136(02)00060-2).

494 [39] Designation: D7334 – 08 (Reapproved 2013) Standard Practice for Surface Wettability of Coatings,
495 Substrates and Pigments by Advancing Contact Angle Measurement 1 n.d.
496 <https://doi.org/10.1520/D7334-08R13>.

497 [40] Standard Test Method for Apparent Shear Strength of Single-Lap-Joint Adhesively Bonded Metal
498 Specimens by Tension Loading (Metal-to-Metal) 1 2019. <https://doi.org/10.1520/D1002-10R19>.

499 [41] Arenas JM, Narbón JJ, Alía C. Optimum adhesive thickness in structural adhesives joints using
500 statistical techniques based on Weibull distribution. *Int J Adhes Adhes* 2010;30:160–5.
501 <https://doi.org/10.1016/j.ijadhadh.2009.12.003>.

502 [42] Gao E, Lu W, Xu Z. Strength loss of carbon nanotube fibers explained in a three-level hierarchical
503 model. *Carbon N Y* 2018;138:134–42. <https://doi.org/10.1016/j.carbon.2018.05.052>.

504 **Figure captions**

505 Fig. 1. The detailed specimen configuration (Unit: mm)

506 Fig. 2. The SLS test setup

507 Fig. 3. SEM images of steel substrates of four roughness levels at a magnification of 100X: (a) the smooth; (b)
508 fine; (c) medium; and (d) coarse substrates

509 Fig. 4. Contact angles of water droplets on four different roughness substrates: (a) the smooth; (b) fine; (c)
510 medium; and (d) coarse substrates

511 Fig. 5 Average stress-strain curves with different surface roughness: (a) the pure; and (b) CNT-reinforced epoxy
512 coatings

513 Fig. 6 Increments between the pure and the CNT-reinforced epoxy coatings: (a) bonding strength; (b) ultimate
514 strain; and (c) toughness

515 Fig. 7 Typical fracture surfaces on smooth substrates: (a) the pure; and (b) CNT-reinforced epoxy coating;

516 Fig. 8. Contact angles of pure and CNT-reinforced epoxy droplets: (a) pure epoxy on the smooth substrate; (b)
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520 Fig. 9 Typical fracture surfaces on coarse substrates: (a) pure epoxy coating; (b) SEM image of the fractured
521 pure epoxy coating at a magnification of 500X; (c) the CNT-reinforced epoxy coating; (d) SEM image of the
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523 Fig. 10 Detection of CNTs: (a) on fracture surfaces at a magnification of 5000X; (b) on fracture surfaces at a
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527 surface roughness

528 Fig. 12 Typical fracture surfaces of the CNT-reinforced epoxy coating on top surfaces at a magnification of
529 500X: (a) on the smooth substrate; (b) on the coarse-blasted substrate;

530

Table 1. The SLS test matrix

Testing group	Specimen quantity	Surface roughness	Coating thickness	Coating material
SE	5	Smooth	0.5 mm	Epoxy
FE	5	Fine	0.5 mm	Epoxy
ME	5	Medium	0.5 mm	Epoxy
CE	5	Coarse	0.5 mm	Epoxy
SC	5	Smooth	0.5 mm	CNT-reinforced epoxy
FC	5	Fine	0.5 mm	CNT-reinforced epoxy
MC	5	Medium	0.5 mm	CNT-reinforced epoxy
CC	5	Coarse	0.5 mm	CNT-reinforced epoxy

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Table 2. Average measured roughness results for four different surface treatments

Roughness level	Grit size	R_z (μm)	STD (%)	R_a (μm)	STD (%)	R_t (μm)	STD (%)
Smooth	60, 120, 240, 400	1.801 ± 0.014	7.77	0.231 ± 0.017	7.36	2.044 ± 0.204	9.98
Fine	60	23.72 ± 1.76	7.42	3.528 ± 0.164	4.64	27.17 ± 2.48	9.13
Medium	36	36.13 ± 0.19	0.53	5.272 ± 0.343	6.51	35.51 ± 1.98	5.58
Coarse	20	46.07 ± 4.17	9.05	8.457 ± 0.737	8.71	48.28 ± 2.91	6.03

534

535

Table 3. The SLS test results

Group	Surface roughness, R_a (μm)	Bonding strength (MPa)	STD (%)	Ultimate strain	STD (%)
SE	0.231	9.455 \pm 0.699	7.39	0.534 \pm 0.042	7.87
FE	3.528	12.234 \pm 0.765	6.25	0.692 \pm 0.059	8.53
ME	5.272	14.764 \pm 0.801	5.43	0.766 \pm 0.068	8.88
CE	8.457	15.145 \pm 0.776	5.12	0.805 \pm 0.081	10.00
SC	0.231	11.595 \pm 0.204	1.76	0.607 \pm 0.031	5.11
FC	3.528	18.573 \pm 0.409	2.20	1.416 \pm 0.042	2.97
MC	5.272	19.475 \pm 0.545	2.80	1.702 \pm 0.100	5.88
CC	8.457	20.057 \pm 1.053	5.25	1.876 \pm 0.117	6.24

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537

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Table 4. Parameter values for each curve

	Parameter value: a	Parameter value: Xc	Parameter value: k	R2
Strength	19.992	-0.244	0.679	0.999
Strain	1.928	1.656	0.546	0.999

539

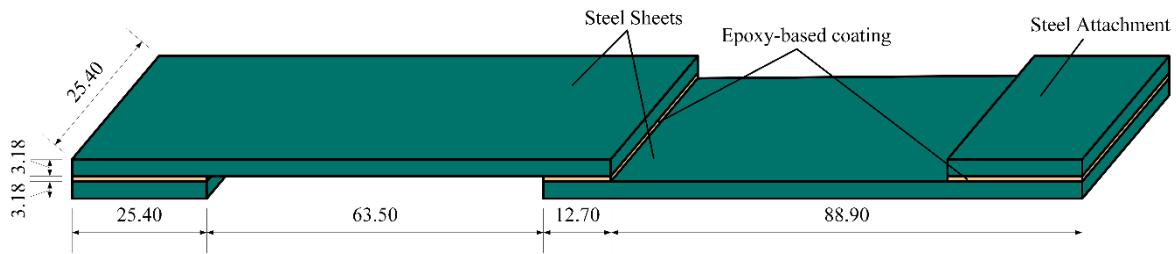
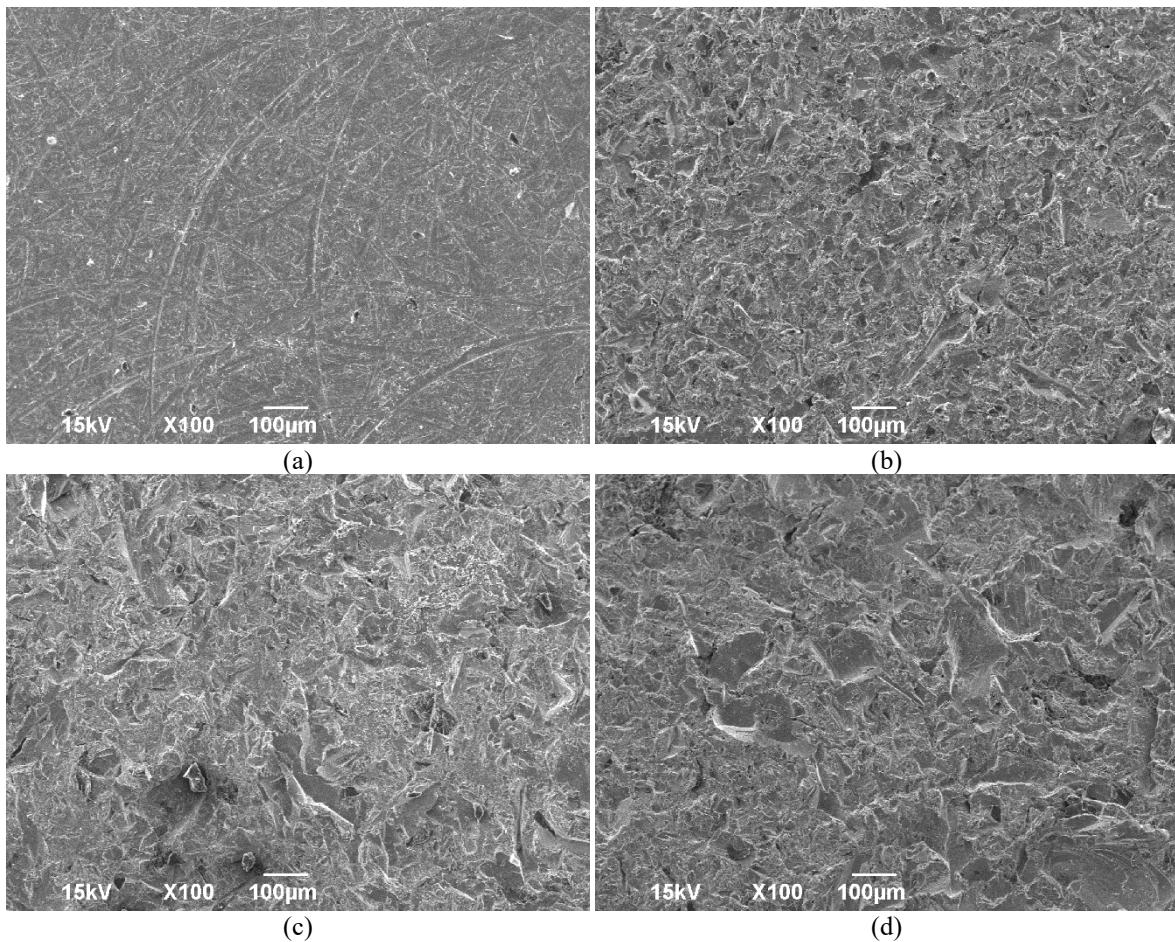


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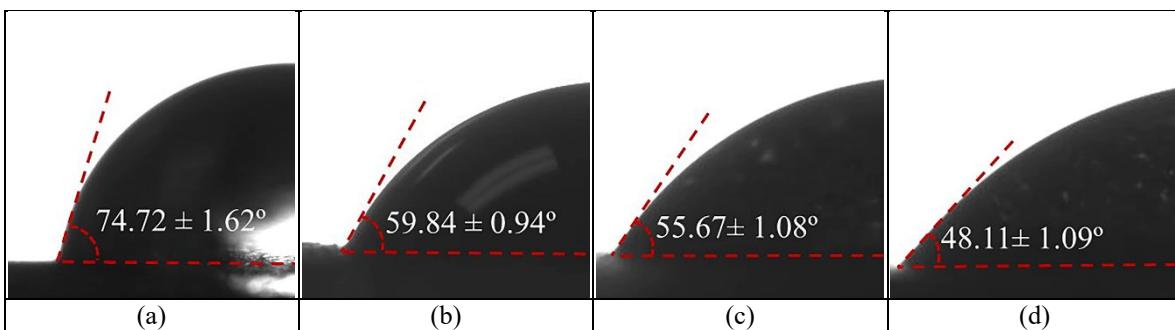


Fig. 2. The SLS test setup

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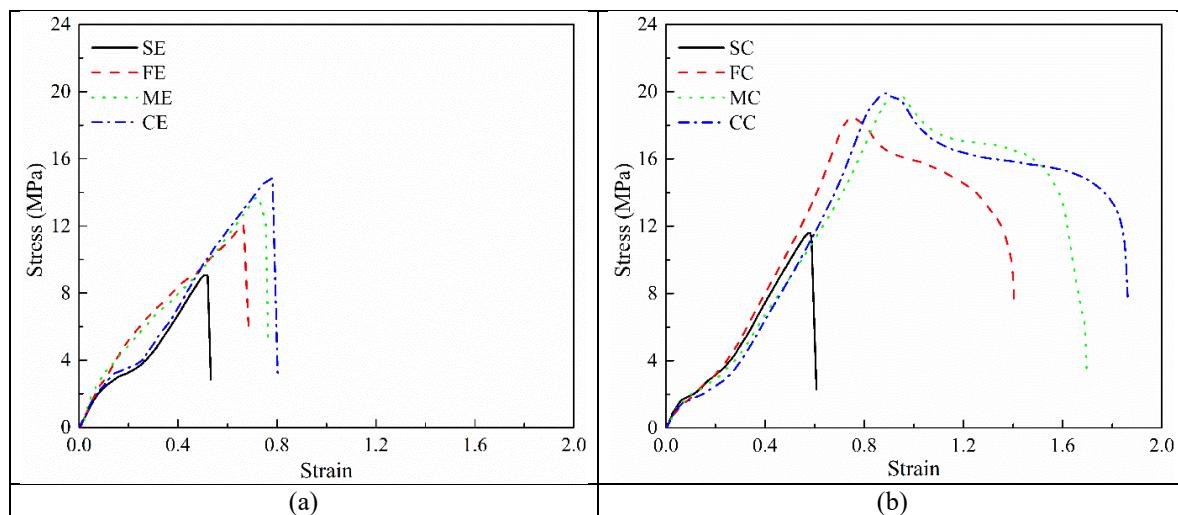


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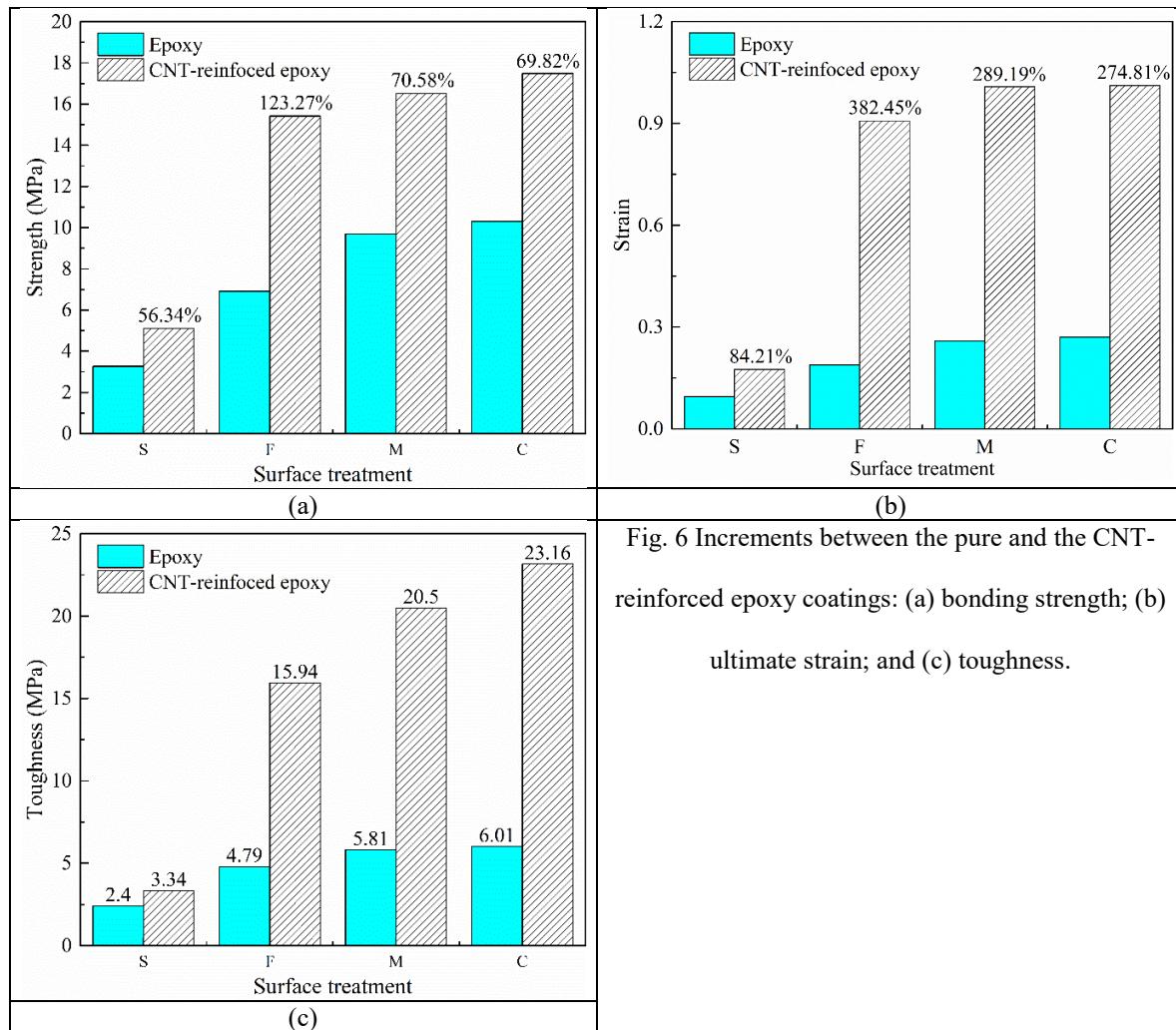
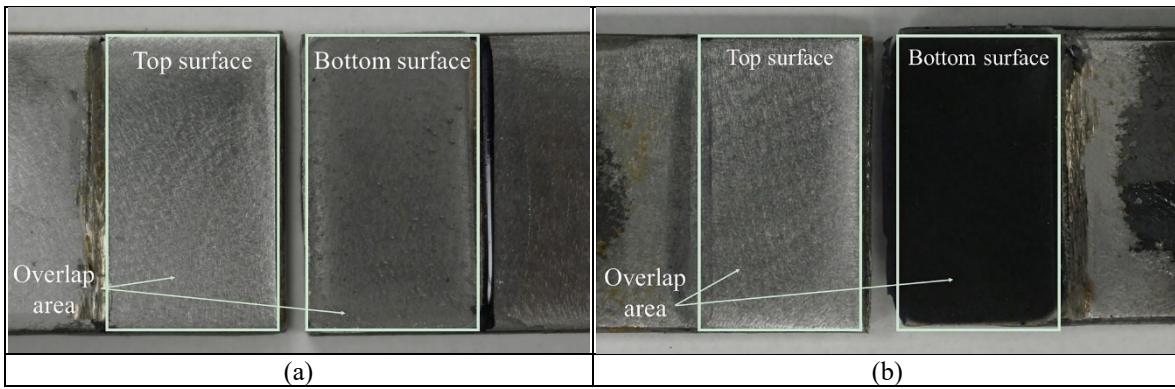
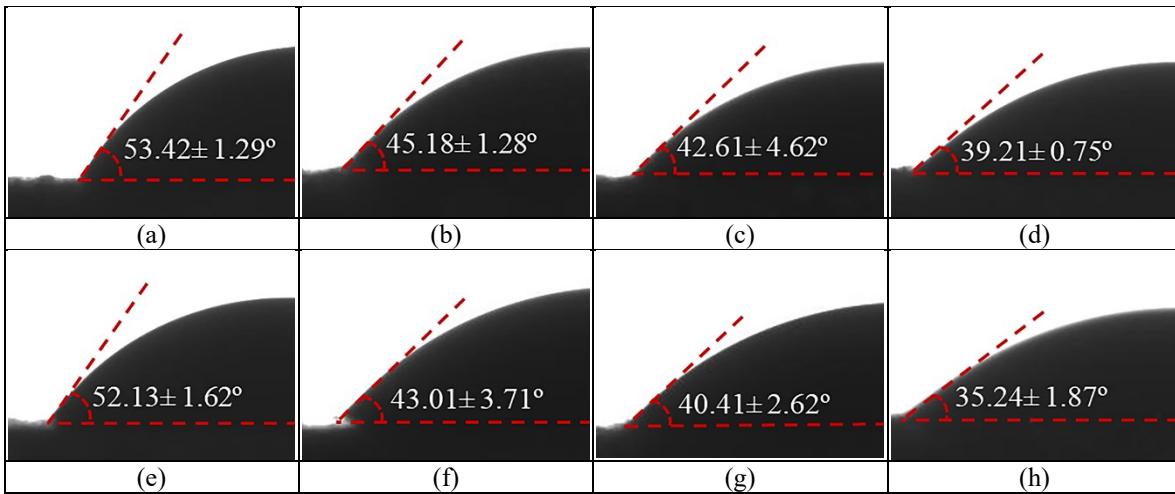


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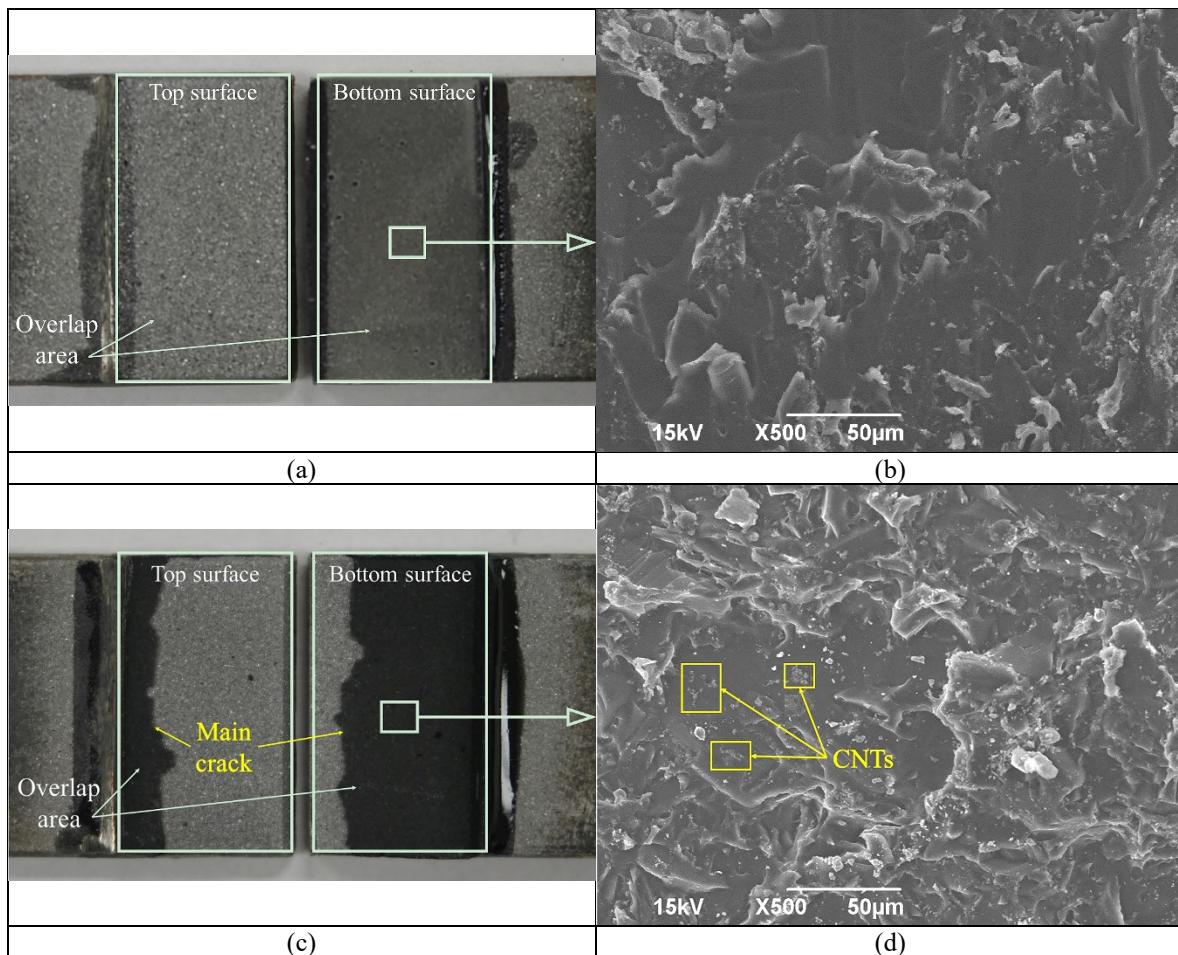


552 Fig. 7 Typical fracture surfaces on smooth substrates: (a) the pure; and (b) the CNT-reinforced epoxy coating.



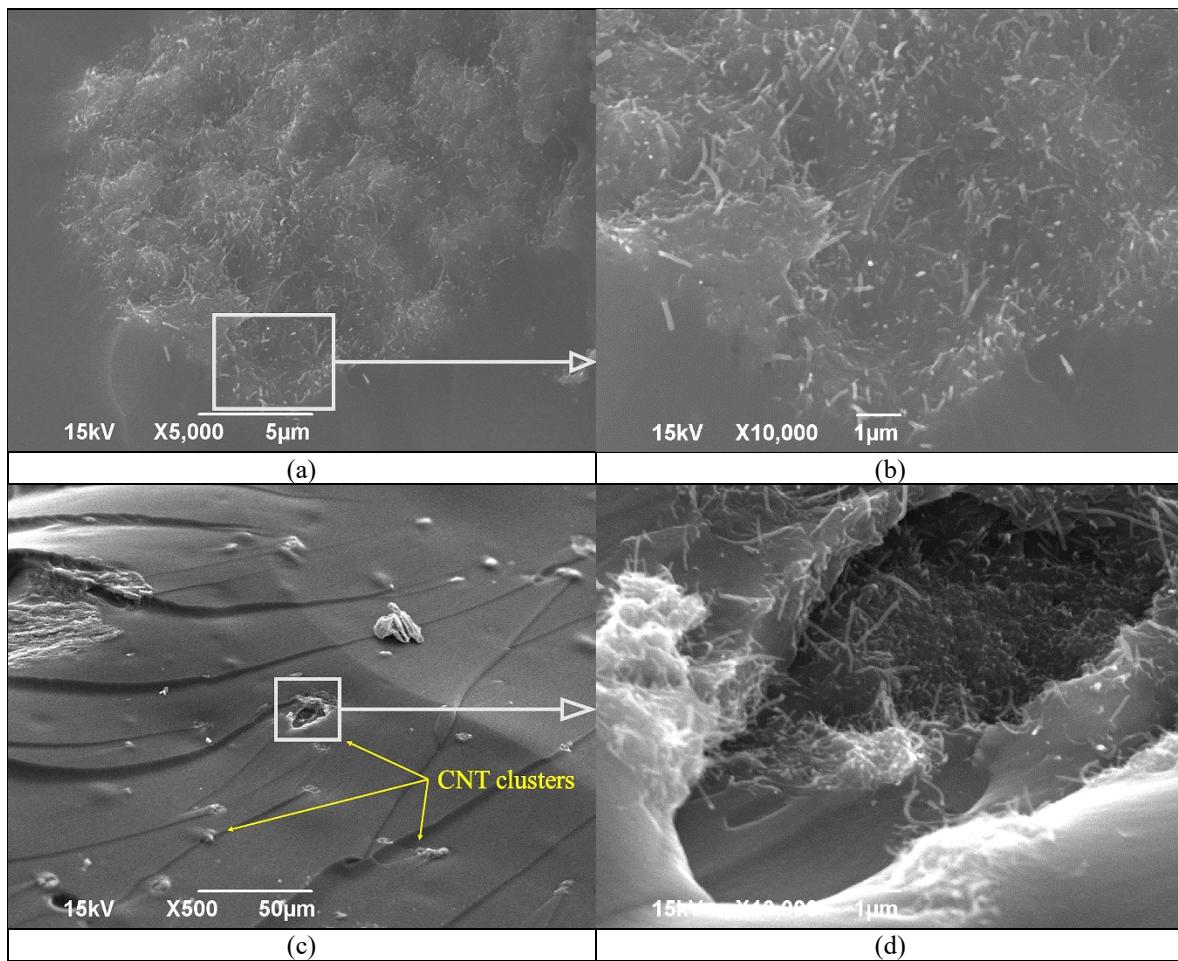
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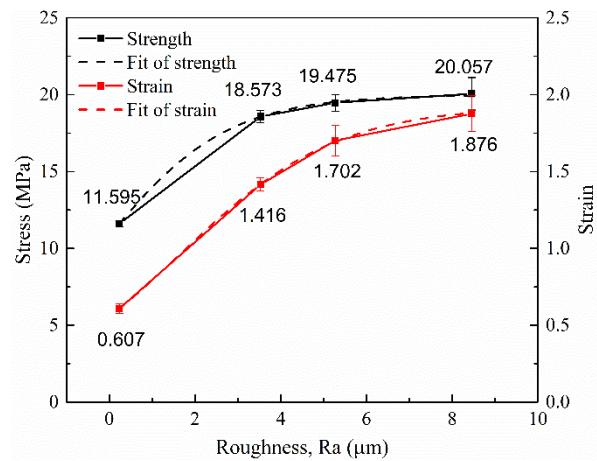


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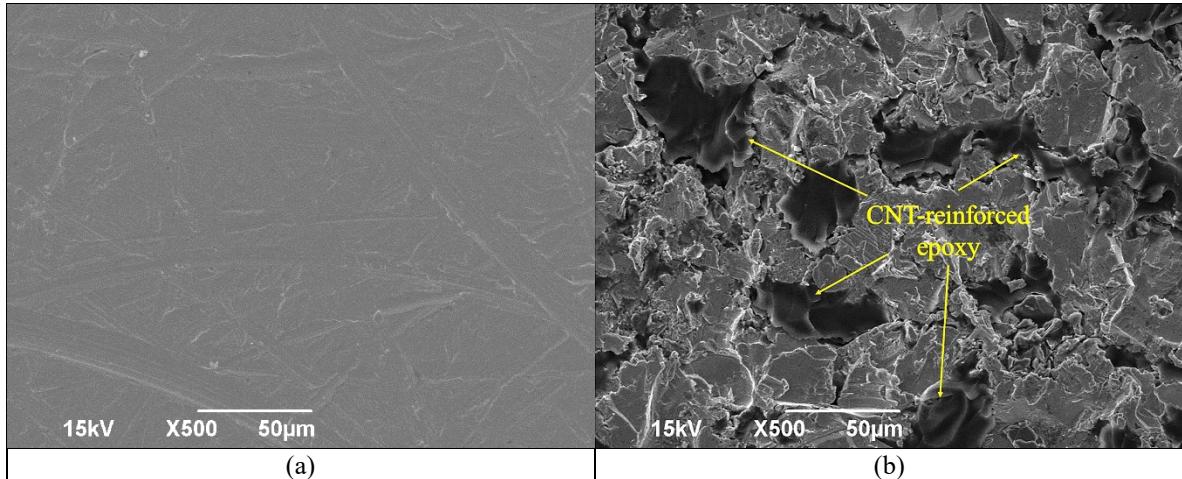


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