Structural Monoclinicity and Its Coupling to Layered Magnetism in Few-Layer CrI₃

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Abstract: Using polarization-resolved Raman spectroscopy, we investigate layer number, temperature, and magnetic field dependence of Raman spectra in one- to four-layer CrI₃. Layer-number-dependent Raman spectra show that in the paramagnetic phase a doubly degenerated E_g mode of monolayer CrI₃ splits into one A_g and one B_g mode in *N*-layer (N > 1) CrI₃ due to the monoclinic stacking. Their energy separation increases in thicker samples until an eventual saturation. Temperature-dependent measurements further show that the split modes tend to merge upon cooling but remain separated until 10 K, indicating a failed attempt of the monoclinic-to-rhombohedral structural phase transition that is present in the bulk crystal. Magnetic-field-dependent measurements reveal an additional monoclinic distortion across the magnetic-field-induced layered antiferromagnetism-to-ferromagnetism phase transition. We propose a structural change that consists of both a lateral sliding toward the rhombohedral stacking and a decrease in the interlayer distance to explain our experimental observations.

The physical properties of two-dimensional (2D) van der Waals (vdW) materials can be drastically influenced by their interlayer electronic and magnetic coupling, which is sensitive to many factors such as stacking symmetry,¹⁻⁵ interlayer angular twist and vertical spacing,⁶⁻⁹ as well as external stimuli.¹⁰⁻¹⁵ Recently, the van der Waals magnet CrI₃ represents an intriguing case in which the stacking symmetry plays a key role in defining the magnetic ground states.^{1, 11, 12, 16} Bulk CrI₃ has a monoclinic stacking at room temperature and undergoes a structural phase transition to rhombohedral stacking near $T_s \approx 220$ K,¹⁷ whereas exfoliated few-layer CrI₃ retains a monoclinic stacking through all temperatures.¹⁸ Theoretical calculations have shown that the rhombohedral and monoclinic stackings favor ferromagnetic (FM) and antiferromagnetic (AFM) interlayer exchange coupling, respectively.¹ Accordingly, bulk CrI₃ exhibits an FM order below the Curie temperature (T_c) of ~ 61 K,¹⁷ while few-layer CrI₃ develops a layered AFM order below the magnetic onset ($T_{\rm M}$) of ~ 45 K.^{19,20} Recent magnetic force microscopy²¹ and Raman spectroscopy²² studies further show that in bulk CrI₃ the surface layers up to a few tens of layers do exhibit layered AFM order, which is possibly attributed to the persistence of monoclinic stacking in the surface. A careful layer-number-dependent study of stacking symmetry and its coupling with magnetism is thus important for understanding the distinct behaviors between surface layers and the deep bulk of 3D CrI₃, as well as those in 2D CrI₃, which has however remained missing until now.

The intimate coupling between the layered AFM-to-FM and structural monoclinic-torhombohedral phase transitions can be controlled by external stimuli. It has been shown that hydrostatic pressure can switch the layered AFM to FM state *via* tuning the layer stacking in bilayer CrI_3 .^{11, 12} While these studies focus on the role of structure in determining the magnetism, an equally important question, that is, how the structure responds to the change of magnetic states, including both the temperature-driven paramagnetic (PM) to layered AFM and the magnetic-fieldinduced layered AFM to FM phase transitions, is yet to be studied. The relationship between stacking and magnetism is, in fact, a more generic topic to many 2D magnets than just specific to CrI_3 . In particular, the correlation between stacking symmetry and magnetic structure has also been reported in another two chromium trihalides, $CrBr_3^{4, 23}$ and $CrCI_3$.²⁴ Despite some earlier work in relatively thick CrI_3 flakes,^{18, 22, 25, 26} a comprehensive and thorough layer-number-dependent investigation of stacking symmetries ranging from mono- to few-layer CrX_3 (X = Cl, Br, and I) remains missing; so does the structure response to magnetic phase transitions, including both the temperature-driven and the magnetic-field-induced ones.

In this work, we carry out careful layer-number-, temperature-, and magnetic-field-dependent polarized Raman spectroscopy measurements (see experimental details in the Methods section). In contrast to the existing Raman works ^{13, 20, 22, 27-29} which are devoted to to the magneto-Raman effect of the A_g phonon mode at ~129 cm⁻¹, we focus on a doubly degenerate E_g phonon mode at $\sim 107 \text{ cm}^{-1.30}$ (see Fig. 1a) defined in the D_{3d} point group in monolayer (1L) CrI₃, which has been demonstrated to be sensitive to interlayer stacking in an angle-resolved polarized Raman spectroscopy study.^{11, 12, 18} We start with layer-number dependent measurements in the PM phase. Figure 1b shows the Raman spectra of one- to four-layer (1-4L) CrI₃ in the frequency range of 95 - 120 cm⁻¹ in both linearly parallel and crossed channels at 80 K (greater than the magnetic onset $T_{\rm M}$ = 45 K). For 1L CrI₃, we observe a doubly degenerate E_g mode at 107 cm⁻¹ in both parallel and crossed channels with an equal intensity. In contrast, for 2-4L samples, the Eg mode of 1L CrI3 splits into two modes. The variations of the relative intensity between the two modes among samples can be attributed to the incident linear polarization aligning differently with the in-plane crystal axes of 1-4L CrI₃. We can first rule out the possibility of Davydov splitting, for which one would expect a monotonic increase in the number of modes as a function of layer number, as opposed to always two modes for 2-4L CrI3 in Fig. 1b. In particular, in 2L CrI3, which we take as an example here, the lattice structure has inversion symmetry in the PM phase.^{20, 27, 31} Under the assumption that there is no degeneracy lift by the interlayer stacking, the Davydov split is expected to lead to two doubly degenerated modes, among which the parity even mode (E_g) is Raman active, whereas the parity odd mode (E_u) should be Raman silent, resulting in only one mode detected in Raman spectra, which is in contrast to our observation of two modes in 2L CrI₃. As a result, we can attribute the origin of split modes only to the monoclinic stacking, which lifts the degeneracy of the $E_g(D_{3d})$ mode in 1L CrI₃, resulting in two nondegenerate modes, one $A_g(C_{2h})$ and one $B_g(C_{2h})$, in *N*-layer $\operatorname{CrI}_3(N > 1)$.

We then fit the spectra to a double-Lorentzian profile in the form of $\sum_{i=1}^{N=2} \frac{A_i \left(\frac{\Gamma_i}{2}\right)^2}{(\omega - \omega_i)^2 + \left(\frac{\Gamma_i}{2}\right)^2} + C$,

where ω_i is the central frequency, Γ_i is the line width, A_i is the peak intensity, and C is a constant

background. Figure 1c shows the frequencies of the fitted modes as a function of layer number N. The energy separation between the two split modes grows bigger at increasing N and eventually saturates at N = 13, indicating an enhanced interlayer coupling in thicker samples. To quantitatively evaluate the evolution of the monoclinic coupling with the layer number, we propose a minimal model based on an array of coupled harmonic oscillators:

$$H = H_0 + \sum_{\langle ij \rangle} \left[a(x_i - x_j)^2 + bx_i x_j + c(y_i - y_j)^2 + dy_i y_j \right]$$

with $H_0 = \frac{1}{2} m \sum_i (\dot{x}_i^2 + \dot{y}_i^2) + \frac{1}{2} k \sum_i (x_i^2 + y_i^2).$

 H_0 represents the E_g mode at a frequency of $\omega_0 = \sqrt{\frac{k}{m}}$ within each layer, where *m* and *k* are the effective mass and spring constant. x_i and y_i are the magnitude of the E_g mode atomic displacement in the *i*th layer, and $\langle ij \rangle$ indicates only the coupling between the nearest neighboring layers. We note that the $a(x_i - x_j)^2$ and $c(y_i - y_j)^2$ terms are typically included in a conventional linear chain model,²⁷ but they do not lead to the splitting of the two observed Raman modes. To account for the monoclinic stacking, we modify the model by adding the bx_ix_j and dy_iy_j terms, which are allowed by the monoclinic C_{2h} symmetry. Diagonalizing *H* leads to 2*N* nondegenerate eigenfrequencies in *N*-layer CrI₃, and the modes with eigenvectors of $(1,0,1,0\cdots 1,0)$ and $(0,1,0,1\cdots 0,1)$ under the $b \to 0$ and $d \to 0$ limit correspond to the two split modes we observe. Our calculations show that the energy separation is linearly proportional to the difference between *b* and *d*, *i.e.*, $\Delta \omega = f|b - d|$, and the coefficient *f* saturates as *N* increases. As shown in Fig. 1d, by setting $|b - d| = 0.79 \ cm^{-1}$ and all other parameters to be unity, our modified linear chain model can reproduce our experimental data well (see Supplementary Section 1 for detailed calculations).

To pin down the Raman tensors for the two split modes in *N*-layer CrI₃ with monoclinic stacking (N > 1), we further carry out Raman selection rule measurements on 2L CrI₃ under the circular polarization basis. As shown in Fig. 1e, the split A_g/B_g modes observed in the linear polarization channels dominantly appear in the circularly crossed channels (LR and RL), while the signal in the circularly parallel channels (LL and RR) is negligible, where L/R(R/L) stands for the polarization channel with the left/right (right/left)-handed circular polarization for the incident (scattered) light. Only one broad single peak is observed in the LR/RL channel because the A_g and

 B_g modes are spectrally too close to be resolved when they appear in the same channel. These observations allow us to write the Raman tensors as

$$E_g = \begin{pmatrix} m & g \\ g & -m \end{pmatrix}, \ A_g = \begin{pmatrix} m & 0 \\ 0 & -m \end{pmatrix}, \ B_g = \begin{pmatrix} 0 & g \\ g & 0 \end{pmatrix},$$

so that the joint tensor $A_g+B_g=\begin{pmatrix} m & g \\ g & -m \end{pmatrix}$ satisfies the selection rules shown in Fig. 1e. It is worth noting that the Raman tensor of the A_g mode has only one independent element in comparison to the generic form of the A_g mode in the C_{2h} point group $\begin{pmatrix} m & 0 \\ 0 & n \end{pmatrix}$, which corresponds to a specific configuration of monoclinic stacking that constrains n = -m.

Having established the characteristics of the monoclinic stacking in *N*-layer CrI₃, we are ready to investigate this structural response of monoclinic stacking to magnetic phase transitions. We use 2L CrI₃ as a prototypical example in the following studies unless otherwise specified. We first carry out temperature-dependent measurements in linear polarization channels over a wide range from room temperature to 10 K. The crystal axis is carefully aligned so that the split modes appear separately in the parallel and crossed channels. As shown in Fig. 2a, the two split modes both exhibit a significant blue shift upon cooling. It is found that the energy separation between the two modes decreases in this process (see Fig. 2b). Unlike bulk CrI₃, where the Ag and Bg modes eventually merge across the monoclinic-to-rhombohedral structural phase transition,¹⁸ in 2L CrI₃, despite a tendency toward rhombohedral stacking, the finite separation between the split modes, although small, demonstrates that the monoclinic stacking persists down to 10 K. Moreover, the absence of any abrupt change in the energy separation across $T_{M, c, s}$ indicates that the structural evolution in this process follows a gradual manner. However, as shown in Fig. 2c, the Raman intensities of the split modes suddenly decrease below T_M , which originates from the spin-phonon-coupled scattering mechanism.²⁷

We then carry out magnetic-field-dependent measurements in circular polarization channels at 10 K. It is known that 2L CrI₃ undergoes a layered AFM-to-FM phase transition at a critical field (B_c) around ± 0.6 T.¹⁹ Figure 3a shows the Raman spectra acquired in LL and RR channels in the frequency range of 95 - 120 cm⁻¹ at 0 and ± 2 T. At 0 T, in contrast to the case of the PM phase (Fig. 1e), the signal of the joint A_g/B_g mode at 107 cm⁻¹ (circle) is enhanced in the layered AFM state. Moreover, two additional modes emerge at ~104 cm⁻¹ (square) and ~115 cm⁻¹ (diamond), and they obey the same selection rule as the 107 cm⁻¹ mode, indicating these two modes are also derived from the joint A_g/B_g mode. These observations allow us to write their Raman tensor as

$$A_g + B_g(AFM) = \begin{pmatrix} m & g + hi \\ g - hi & -m \end{pmatrix},$$

where the antisymmetric terms break the time-reversal symmetry and account for the layered AFM-assisted phonon scattering.²⁷ By applying an out-of-plane magnetic field (B_{\perp}) of ± 2 T, a Raman circular dichroism emerges for all these three modes, with opposite relative strengths in the LL/RR channels for opposite magnetic field directions. Shown in Fig. 3b is the fitted Raman intensity of the three modes as a function of B_{\perp} . Clearly, the emergence of the Raman circular dichroism happens at the critical magnetic field $B_c = \pm 0.6$ T. Such deviation of the selection rule in the magnetic-field-induced FM phase can be attributed to the change in the Raman tensor:

$$A_g + B_g(FM) = \begin{pmatrix} m' & g' + h' \mathbb{i} \\ g' - h' \mathbb{i} & n' \end{pmatrix}.$$

The key finding here is that the constraint in the diagonal tensor elements (*i.e.*, n' = -m') is lifted, and one more independent tensor element is introduced. As a result, the structural contribution in the $A_g + B_g(FM)$ Raman tensor, *i.e.*, the real part, matches the generic form of the monoclinic C_{2h} case, and thus an additional monoclinic structural distortion is detected across this magnetic-fieldinduced layered AFM-to-FM phase transition.

We then look into the energy separation between the two split modes in the linearly polarized channels to further evaluate this monoclinic distortion across B_c . Figure 4a shows the Raman spectra in linear polarization channels at selected magnetic fields (a different sample from that in Fig. 2). The A_g/B_g modes can be well resolved in both the linearly parallel and crossed channels at 0 T, while they evolve to nearly merge at 1 T and above. Figures 4b and 4c show the fitted frequencies of the two split modes and their separation as a function of B_{\perp} , respectively. A step-like reduction in the energy separation across B_c is captured, consistent with the first-order nature of the field-induced layered AFM-to-FM transition. We note that the decrease in the doublet frequency separation has also been observed in the temperature-dependent data in Fig. 2b, where upon cooling, a tendency of the monoclinic-to-rhombohedral transition is proposed and a reduction in the interlayer distance was reported previously.¹⁷

On the basis of our observations above, we present the following interpretation for the response of the monoclinic structure to the magnetic phase transitions in N-layer CrI₃. To address the suppression of the monoclinic-to-rhombohedral phase transition in N-layer CrI₃, we carry out density functional theory (DFT) calculations of the total elastic energy as a function of lateral shift between the two layers with respect to the monoclinic stacking. Our results show that there is an energy barrier of ~ 40 meV between the local minimum of monoclinic stacking and the global minimum rhombohedral stacking, which is comparable with the value in the previous report.¹ A plausible explanation is that N-layer CrI₃ is trapped in the monoclinic local minimum at room temperature, and the thermal energy is not sufficient to overcome this energy barrier. Despite the absence of the monoclinic-to-rhombohedral structural phase transition in N-layer CrI₃, the lattice structure still undergoes a lateral sliding toward the rhombohedral structure, as evidenced by the decrease in the energy separation of the split modes upon cooling (Fig. 2). Meanwhile, as reported in bulk CrI₃,¹⁷ we expect the interlayer distance of few-layer CrI₃ also decreases due to thermal contraction. Therefore, we propose that the structural change upon cooling consists of two parts: a lateral sliding toward the rhombohedral stacking and a decrease in the interlayer distance. To reconcile with our modified linear chain model, we comment that first, the values of b and d, which quantify monoclinicity, increase, since the reduced interlayer distance enhances the interlayer coupling. Second, the difference between b and d, which is correlated to the energy separation between the two modes, decreases, as the adjacent layers slide toward the rhombohedral stacking. This is the scenario that happens across the magnetic-field-induced layered AFM to FM transition at B_c , where the reduction in the interlayer distance is responsible for the enhanced interlayer interaction that leads to the emergent Raman circular dichroism and the lateral sliding toward rhombohedral stacking accounts for the decrease of mode frequency separation across B_{c} . Compared to the smooth merging of the two modes upon cooling (Fig. 2b), the adjacent layers experience a more abrupt sliding across B_c that is shown by the sharp step-like drop in the energy separation between the two modes (Fig. 4c).

In summary, we have carried out a comprehensive investigation of the monoclinic stacking of N-layer CrI₃ and tracked its response to two magnetic phase transitions, the spontaneous thermal one and the magnetic-field-induced spin-flip one in 2L CrI₃. A clear monoclinic-stacking-induced degeneracy lift from E_g in monolayer to A_g+B_g in N-layer (N = 2, 3, 4, ...) CrI₃ is demonstrated in

the layer-number-dependent results. On one hand, such a monoclinic structure in *N*-layer CrI_3 exhibits a failed attempt to transit into the rhombohedral structure upon cooling. On the other hand, this monoclinic structure undergoes a further monoclinic distortion across the magnetic-field-induced layered AFM-to-FM transition. Our results show that the crystal structure also responds intimately to the magnetic phase transition in *N*-layer CrI_3 , complementary to the reported magnetic phase transition following a change in the stacking symmetry,^{11, 12} establishing the presence of magnetoelastic coupling in *N*-layer CrI_3 and suggesting a two-way control between structure and magnetism, *i.e.*, changing structure (magnetism) *via* controlling magnetism (structure).

Methods/Experimental

Sample Fabrication. CrI₃ single crystals were grown by the chemical vapor transport method, as detailed in ref. [20, 22, 27, 32]. The CrI₃ samples were exfoliated in a nitrogen-filled glovebox. Using a polymer-stamping transfer technique inside the glovebox, we sandwiched CrI₃ flakes between two few-layer hBN flakes and transferred them onto SiO₂/Si substrates for Raman spectroscopy measurements.

Raman Spectroscopy. Micro-Raman spectroscopy measurements were carried out using a 633nm excitation laser. The incident beam was focused by a 40× objective down to ~3 μ m in diameter at the sample site, and the power was kept at ~80 μ W. The scattered light was collected by the objective in a backscattering geometry, then dispersed by a Horiba LabRAM HR Evolution Raman spectrometer, and finally detected by a thermoelectric-cooled CCD camera. A closed-cycle helium cryostat is interfaced with the Raman system for the temperature-dependent measurements. All thermal cycles were performed at a base pressure that is lower than 7 × 10⁻⁷ mbar. In addition, a cryogen-free magnet is integrated with the low-temperature cryostat for the magnetic-field-dependent measurements. In this experiment, the magnetic field was applied along the out-of-plane direction and covered a range of 0 to 2.2 T.

Density Functional Theory Calculations.

Density-functional theory calculations of monoclinic few-layer CrI₃ were performed using the projected augmented-wave (PAW) method as implemented in the Vienna *Ab Initio* Simulation

Package (VASP).^{33, 34} In all calculations, we adopted the Perdew–Burke–Ernzerhof (PBE) exchange-correlation functionals, with valence electron configurations as $3d^54s^1$ and $5s^25p^5$ for Cr and I, respectively. van der Waals corrections with optB86b-vdW flavor^{35, 36} have been applied to account for the van der Waals interlayer interactions. The Brillouin zone was sampled by a $9 \times 9 \times 1$ Monkhorst-Pack k-point grid mesh, and a 500 eV plane-wave cutoff energy was used. Relaxations were performed until the Hellmann-Feynman force on each atom became smaller than 0.002 eV/Å and the total energy was converged to be within 10^{-6} eV. Following intralayer ferromagnetic ordering, magnetic moments for all Cr atoms in the same layer were initialized in the same direction.

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Fig. 1. a. Atomic displacement of the E_g mode (107 cm⁻¹) in monolayer Crl₃. **b.** Raman spectra of 1-4L Crl₃ acquired at 80 K in the linearly parallel and crossed polarization channels. The single E_g mode in 1L Crl₃ is fitted to a single-Lorentzian profile, and the split A_g and B_g modes in 2-4L Crl₃ are fitted to double-Lorentzian profiles. **c.** Plot of the frequencies of the split modes identified in **b** as a function of layer number (*N*). Data from 5L,9L and 13L samples are included for comparison. **d.** Energy separation ($\Delta\omega$) of the split modes as a function of *N*. Solid squares are experimental data, and the solid curve is a fit to the linear chain model. **e.** Circular polarization selection rules of Raman spectra of 2L Crl₃ acquired at 80 K. Error bar stands for one standard error of the fitting parameter.

Figure 2



Fig. 2. a. Raman spectra (dots are raw data and solid curves are single-Lorentzian fits) of 2L Crl₃ acquired at selected temperatures in the linearly parallel and crossed polarization channels. Spectra are vertically offset for clarity. **b.** Plot of the energy separation between the split modes as a function of temperature. **c.** Plots of the intensities of the split modes as a function of temperature. Vertical dashed line in **b** and **c** marks the magnetic transition temperature $T_{\rm M}$ = 45 K. Error bar stands for one standard error of the fitting parameter.





Fig. 3. a. Raman spectra of 2L Crl₃ acquired at 10 K in the cocircularly polarized LL and RR circular polarization channels with 0 and ± 2 T external magnetic fields. **b.** Plot of the intensities of the three Raman modes labeled as square, circle, and diamond in **a** as a function of external magnetic field (B_{\perp}). Error bar stands for one standard error of the fitting parameter.

Figure 4



Fig. 4. a. Raman spectra of 2L Crl₃ acquired at 10 K in the linearly parallel and crossed channels at selected magnetic fields as labeled. **b.** Plot of the fitted frequencies of the two split modes as a function of external magnetic field (B_{\perp}). **c.** Plot of the frequency separation between the two split modes as a function of external magnetic field (B_{\perp}). Error bar stands for one standard error of the fitting parameter.

Supplementary Information for

Structural Monoclinicity and Its Coupling to Layered Magnetism in Few-Layer CrI₃

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S2. Density functional theory calculation of bilayer CrI₃

S1. Linear chain model

As described in the main text, we propose a modified linear chain model in the simplest form of

$$H = H_0 + \sum_{\langle ij \rangle} \left[a (x_i - x_j)^2 + b x_i x_j + c (y_i - y_j)^2 + d y_i y_j \right]$$

with $H_0 = \frac{1}{2} m \sum_i (\dot{x}_i^2 + \dot{y}_i^2) + \frac{1}{2} k \sum_i (x_i^2 + y_i^2).$

 H_0 represents the E_g mode at a frequency of $\omega_0 = \sqrt{\frac{k}{m}}$ within each layer, where *m* and *k* are the effective mass and spring constant. x_i and y_i are the magnitude of the E_g mode atomic displacement in the *i*th layer, and $\langle ij \rangle$ indicates only the coupling between the nearest neighboring layers. The two terms in H_0 describe the kinetic energy and elastic potential energy of each atomic layer, respectively, and the rest are responsible for the interaction between neighboring layers. The Raman modes we observe in the Raman spectra correspond to the homogeneous A_g and B_g eigenmodes $(x_1, y_1, x_2, y_2 \dots, x_N, y_N) = (1,0,1,0,\dots,1,0)$ and $(0,1,0,1,\dots,0,1)$ under the $b \to 0$ and $d \to 0$ limit, where N is the total layer number. In the Hamiltonian accounting for the interlayer coupling, the $a(x_i - x_j)^2$ and $c(y_i - y_j)^2$ terms represent the energy cost induced by the relative displacement between adjacent layers. In the conventional linear chain model where b = d = 0, the frequencies of the homogeneous A_g and B_g modes are solved to be $\omega_1 = \omega_2 = \omega_0$ with no energy splitting.

To reconcile with our experimental finding of the two split modes in *N*-layer CrI₃, we introduce the minimally required perturbative corrections bx_ix_j and dy_iy_j ($b \ll a$ and $d \ll c$) which are allowed by the C_{2h} symmetry. We can in principle add more terms that obey the C_{2h} symmetry, but simultaneously will introduce redundant fitting parameters. Here, we decide to choose this minimal modification from the conventional linear chain model and show that it is sufficient in explain our results. Physically, these two terms account for the monoclinic coupling between adjacent layers. By diagonalizing the Hamiltonian, we find that this introduction of monoclinicity leads to the splitting of the A_g and B_g modes. The calculated frequencies ω_1 and ω_2 for layer number $N = 1 \sim 4$ are shown in Table S1. Under the assumption of $b \ll a$ and $d \ll c$, we perform Taylor expansion and find that the energy separation between the two modes is linearly proportional to |b - d|. By setting $|b - d| = 0.79 \ cm^{-1}$ and all other parameters to be unity, we

plot the energy separation predicted by the linear chain model, together with the experiment results in Fig. 1d in the main text, which shows an excellent agreement between the two.

N	ω_1^2	ω_2^2
1	ω_0^2	ω_0^2
2	$\omega_0^2 + \frac{b}{m}$	$\omega_0^2 + \frac{d}{m}$
3	$\omega_0^2 + \frac{3a - \sqrt{9a^2 - 8ab + 2b^2}}{m}$	$\omega_0^2 + \frac{3c - \sqrt{9c^2 - 8cd + 2d^2}}{m}$
4	$\omega_0^2 + \frac{4a + b - \sqrt{16a^2 - 16ab + 5b^2}}{m}$	$\omega_0^2 + \frac{4c + d - \sqrt{16c^2 - 16cd + 5d^2}}{m}$

Table S1. Calculated frequencies of the two homogeneous vibration modes ($\omega_{1,2}$) in 1-4L Crl₃ using the modified linear chain model

When the temperature cools down, the energy separation between the two homogeneous A_g and B_g modes decreases which corresponds to a reduction in the difference between *b* and *d*. This indicates a lateral sliding between adjacent layers and one can imagine when the layers slide to the rhombohedral stacking, *b* will equal to *d* and there will be no splitting. The same change may happen across the layered AFM-to-FM phase transition, though the sharp decrease in the energy separation may indicate a more abrupt and fierce sliding compared to the one in the cool down process.



S2. Density functional theory calculation of bilayer CrI₃

Fig. S1 The full energy landscape over the lateral fractional shift under (a) AFM and (c) FM interlayer coupling, respectively. (b) and (d) show a line cut of (a) and (c) along the [110] direction. An energy barrier of ~40 meV is observed between the monoclinic stacking (point A) and rhombohedral stacking (point B).

We have investigated the possible structural phase transitions in monoclinic bilayer CrI₃ (point group symmetry C_{2h}). For both AFM and FM interlayer magnetic orders, we shifted the upper layer with respect to the bottom one with a combined fractional coordinate along the [100] and [110] directions. A total of $16 \times 16 = 256$ configurations has been calculated. For each configuration, we fixed the atomic coordinates in the layer plane while allowing the vertical coordinates to get fully optimized. The energy difference (ΔE) relative to the starting monoclinic structure (denoted as A in Fig. S1b and Fig. S1d) has been calculated and the resulted energy landscapes have been shown in Fig. S1a and Fig. S1c for AFM and FM orders, respectively. The corresponding typical energy differences ΔE along the [110] direction have been shown in Fig. S1b and Fig. S1b and Fig. S1b and Fig. S1d, corresponding to the R $\overline{3}$ rhombohedral stacking order) in both AFM and FM. Such energy barriers might not be easily overcome with the external magnetic field, thus prohibiting a possible structural phase transition.

Note that in the FM order, the energy of B structure will be about -10.5 meV lower than that of A. However, this energy difference becomes much smaller in the AFM order (~1.4 meV).