

## Elucidating Cathodic Corrosion Mechanisms with Operando Electrochemical Liquid-Cell STEM in Multiple Dimensions

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Cathodic corrosion is an enigmatic electrochemical process that corrodes metallic electrodes under sufficiently reducing potentials. First discovered by Fritz Haber in the 19<sup>th</sup> century,<sup>1–3</sup> the dynamic cathodic corrosion mechanisms remain elusive, due to the lack of *operando/in situ* microscopic methods for probing electrified solid-liquid-gas interfaces.<sup>4</sup> These poorly-understood mechanisms also have modern relevance, as they underlie the structural evolution of nanosized electrocatalysts during many electrochemical reactions under reducing potentials, in particular, CO<sub>2</sub> and N<sub>2</sub> reduction for their potential to lower carbon emission and produce valuable chemicals.<sup>5,6</sup>

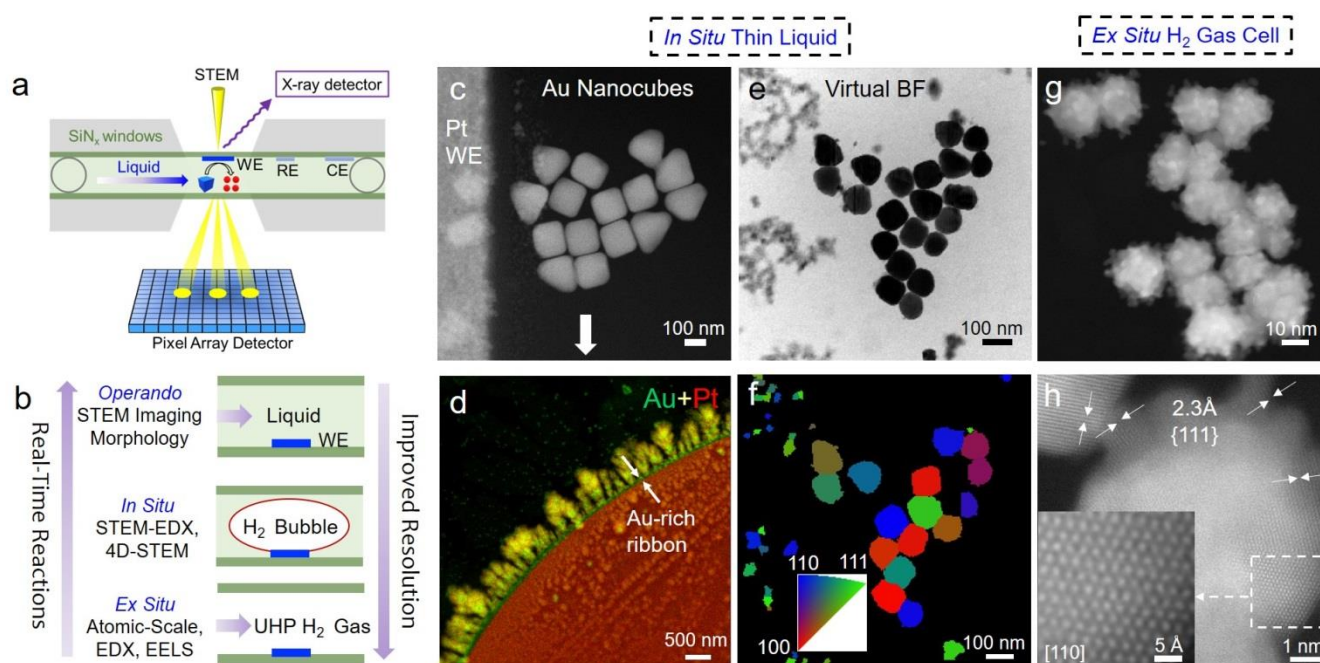
Here, we employed *operando* electrochemical liquid-cell STEM (EC-STEM) to reveal that cathodic corrosion led to a significantly higher level of structural destruction for Au nanocubes than bulk Pt electrodes, accompanied by the unexpected formation of Au-Pt alloy nanostructures, which are thermodynamically immiscible in the Au-Pt phase diagram. Dynamic evolution in morphology, composition, and crystallographic information was retrieved by multidimensional STEM techniques, namely 2D-STEM imaging, 3D STEM-EDX elemental mapping, and 4D-STEM diffraction imaging. We designed a strategy to enable *operando*, *in situ* and *ex situ* EC-STEM operation in the same cell, and at identical locations to enable reliable electrochemical measurements of cathodic corrosion and simultaneously probe real-time interfacial changes across multiple scales (**Figs. 1a,b**) by generating H<sub>2</sub> bubbles to reversibly control the liquid thickness.

*Operando* EC-STEM was used to track the morphological changes in a thick electrolyte under similar conditions to standard electrochemical measurements (**Fig. 2**). The nucleation and formation of H<sub>2</sub> bubble were evidenced by cyclic voltammetry (CV), where well-defined redox peaks progressively increased as the potential limits decreased from -1.0 to -1.8 V, followed by a rapid decay to nearly zero at -2.0 V vs. Pt (**Figs. 2a,b**). A native thinner liquid layer, created by the H<sub>2</sub> bubble, leads to a significantly improved spatial resolution (**Fig. 2c**). *Operando* EC-STEM images in **Figs. 2d–f** show the agglomeration of nanoparticle (NP) clusters (~100 nm, white boxes) and generation of new Au-Pt alloy NPs (~30 nm, red arrows and green boxes).

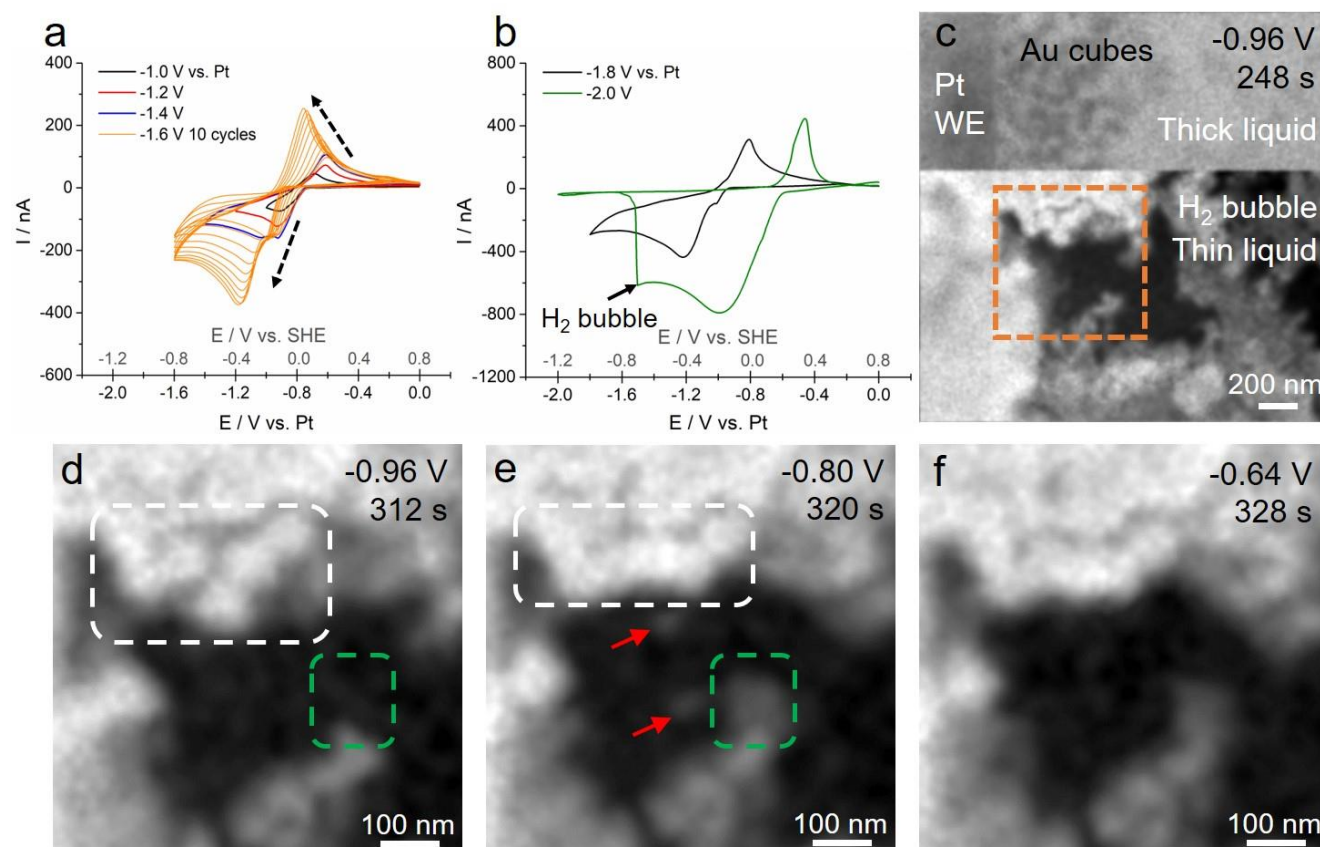
The thinner liquid made *in situ* STEM-EDX and 4D-STEM diffraction imaging possible. STEM-EDX maps show the transformation from Au nanocubes near the bulk Pt electrode (**Fig. 1c**) to Au-rich alloy NPs (~30 nm, Au/Pt 5:1) and Au-Pt alloy nanostructures (Au/Pt 1:1) on the Au-rich ribbon (~100 nm) (**Fig. 1d**). *In situ* scanning electron nanobeam diffraction (SEND), using an electron microscopy pixel array detector (EMPAD), provides complementary information on crystal structures besides morphology

and composition.<sup>7-10</sup> Virtual bright field (BF) image and crystallographic orientation map, obtained by the SEND, show the polycrystalline nature of NP clusters and mossy structures (**Figs. 1e,f**). To resolve atomic-scale lattice structures, H<sub>2</sub> gas was flowed to remove the liquid completely, while preserving the corroded electrode at the same locations in a reducing environment. *Ex situ* STEM images of Au<sub>5</sub>Pt alloy NPs in H<sub>2</sub> gas cell configuration clearly exhibit the “nano-spikes” features on the polycrystalline particle surface (**Fig. 1g**) and the atomic structure of an fcc lattice oriented near the [110] zone axis (**Fig. 1h**).

In summary, this work shows higher cathodic corrosion kinetics for nanocrystals than bulk electrodes, and demonstrate how *operando* EC-STEM provides a platform for microscopic investigation at electrode-electrolyte interfaces.<sup>11</sup>



**Figure 1.** Operando/in situ electrochemical liquid-cell STEM studies of cathodic corrosion. (a) Schematic of EC-STEM with working, reference and counter electrodes (WE, RE and CE), and EDX and 4D-STEM capability in liquid. (b) Overview of methodology of operando/in situ/ex situ EC-STEM. (c) High-angle annular dark-field (HAADF) STEM image of pristine Au nanocubes near the Pt WE in liquid. (d) In situ thin-liquid STEM-EDX mapping of heterogeneous Au-Pt alloy nanostructures after corrosion. (e-f) In situ thin-liquid BF-STEM image and SEND-based crystallographic orientation map of corroded NP clusters and mossy structures, where the color represents the superposition of three basal planes in the inverse pole figure. (g-h) Ex situ atomic-scale HAADF-STEM image of Au<sub>5</sub>Pt alloy NPs in ultrahigh purity (UHP) H<sub>2</sub> gas cell.



**Figure 2.** Figure 2 Operando EC-STEM studies of dynamic morphological changes during cathodic corrosion. (a) CV profiles of Au nanocubes on the Pt WE in EC-STEM in a regular thick electrolyte (500–800 nm). 10 continuous cycles from -1.6 to 0 V (orange profiles) showing the process of accumulating soluble H<sub>2</sub>. (b) CV profiles showing a sharp drop in reduction current as the lower limit reached -2.0 V, indicating the formation of a H<sub>2</sub> bubble. (c) Operando EC-STEM image of the thin liquid layer after the H<sub>2</sub> bubble formation. (d–f) Three continuous STEM image frames showing the dynamic morphological changes during cathodic corrosion.

## References

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