

1                   **Helium Diffusion Systematics Inferred from Continuous Ramped Heating**  
2                   **Analysis of Transantarctic Mountains Apatites Showing Age Overdispersion**

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11                  Transantarctic Mountains

13                  **ABSTRACT**

14                  Application of apatite (U-Th)/He thermochronology has been hindered by incomplete  
15                  understanding of diffusion systematics that leads to the single-grain age dispersion often  
16                  displayed by samples, particularly those from older, slowly cooled settings. We applied  
17                  the continuous ramped heating (CRH) method to an apatite suite from Cathedral Rocks  
18                  in the Transantarctic Mountains (TAM) that have high age dispersion in order to explain  
19                  processes that complicate  $^4\text{He}$  diffusion in apatite. Examining 132 apatite grains from a  
20                  total of six samples, we confirmed earlier apatite (U-Th)/He results showing that  
21                  measured AHe ages have at least three-fold intra-sample dispersion with no obvious  
22                  relationships between ages and effective uranium concentration (eU) or grain size. CRH  
23                  results on these apatites yielded two groups. Those with younger ages, characterized by  
24                  unimodal incremental  $^4\text{He}$  gas-release curves, displayed simple volume diffusion  
25                  behavior. In contrast, grains with older ages generally show complex gas release in the  
26                  form of sharp spikes and/or extended gas-release at high temperatures (i.e.,  $\geq 800^\circ\text{C}$ ).  
27                  Simply-behaved apatites still show considerable age dispersion that exceeds what grain  
28                  size, radiation damage, and analytical uncertainty can explain, but this dispersion appears  
29                  to be related to variations in  $^4\text{He}$  diffusion kinetics. The screened AHe ages from simply-  
30                  behaved younger apatite grains together with kinetic information from these grains  
31                  suggest that the sampled region experienced slow cooling prior to rapid cooling (rock  
32                  exhumation) beginning ca. 35 Ma. This interpretation is consistent with other studies  
33                  indicative of an increase in exhumation rates at this time, possibly related to the initiation  
34                  of glaciation at the Eocene-Oligocene climate transition. An attempt to correct older  
35                  apatite ages by simply removing extraneous gas-release components yielded some ages  
36                  that are too young for the samples' geologic setting, suggesting that the factors that lead  
37                  to complex laboratory release behavior can impact both the expected radiogenic  
38                  component as well as those that are apparently extraneous. From our observations, we  
39                  infer that many apatite grains contain imperfections of varying kinds that contribute  
40                  significantly to kinetic variability beyond that associated with radiation damage and  
41                  conclude that: (1) CRH analysis can serve as a routine screening tool for AHe dating and

42 offers opportunities to reveal first-order kinetic variations; (2) model-dependent age  
43 correction may be possible but would require some means of estimating the broad  
44 proportions of  ${}^4\text{He}$  components incorporated into grains before and after closure to  
45 diffusion, and (3) interpretation of highly dispersed AHe ages requires assessment of  
46 individual-grain diffusion kinetics beyond that predicted by radiation-damage models.

47 **1. Introduction**

48 Following the proposal that apatite (U-Th)/He (AHe) ages could be used as a low-  
49 temperature thermochronometer (Zeitler et al., 1987), advances in pursuing the  
50 fundamental diffusion systematics and kinetics of helium release (Wolf et al., 1996; Farley  
51 et al., 1996; Farley, 2000; Shuster et al., 2006; Flowers et al., 2009) has led apatite (U-  
52 Th)/He thermochronology to become widely used in studies of tectonic and surface  
53 processes (e.g., Reiners et al., 2003; Reiners et al., 2005; Fitzgerald et al., 2006; Flowers  
54 and Farley, 2012; Toraman et al., 2014; Long et al., 2015). However, it has become widely  
55 recognized that interpretation of AHe data is often complicated by intra-sample age  
56 variations (commonly referred to as “excess age dispersion”) that are beyond typical  
57 analytical uncertainties (e.g., Fitzgerald et al., 2006; Flowers and Kelley, 2011; Peyton et  
58 al., 2012; Zeitler et al. 2017a; McDannell et al., 2018). Significant efforts have been made  
59 to explain such age dispersion and to unravel complexities in  ${}^4\text{He}$  diffusion systematics.  
60 Some factors, for example, the presence of U-rich micro-inclusions (Farley, 2002), U and  
61 Th zonation (Meesters and Dunai, 2002; Fitzgerald et al., 2006), and  ${}^4\text{He}$  implantation  
62 (Spiegel et al., 2009; Murray et al., 2014) will complicate He analysis or diffusion  
63 systematics in ways that make it difficult to obtain useful apparent ages (Farley, 2000).  
64 Other effects such as grain size (Reiners and Farley, 2001), broken grains, (Beucher et  
65 al., 2013; Brown et al., 2013), and the way that radiation damage systematically alters He  
66 diffusion kinetics (Shuster et al., 2006; Gautheron et al., 2009; Flowers et al., 2009; Willett

67 et al., 2017) can lead to age dispersion that can be exploited to reveal more information  
68 about thermal history. Slow cooling through or long-term residence within an apatite He  
69 partial retention zone will accentuate age dispersion, often to a considerable degree (e.g.,  
70 Reiners and Farley, 2001; Fitzgerald et al., 2006).

71 Despite these contributions, there are still situations where we still cannot fully explain  
72 commonly observed AHe age dispersion. Applied studies of sample suites from different  
73 geologic settings have found that even using careful sample selection, grain size and  
74 radiation damage can only explain some of the observed dispersion (Zeitler et al., 2017a).  
75 To reduce the probability of overdispersed ages and to understand age dispersion should  
76 this occur, common practices include performing “re-extracts” to evaluate if all He has  
77 been out-gassed, collecting data only from single grains, performing large-n replicate  
78 analyses, and plotting He ages vs. size (radius) and effective uranium [eU] to evaluate  
79 excess dispersion. If excess dispersion occurs, complex data sets can be vexing and  
80 difficult to interpret and use in thermal history modeling, and if a large number of samples  
81 and single-grain analyses are undertaken in order to circumvent such issues, these  
82 approaches are time-consuming and costly.

83 As a result, the thermochronology community is actively working on the challenge that  
84 excess age dispersion presents (e.g., Zeitler et al., 2017b; McDannell et al., 2018). A  
85 possible factor in age dispersion that has attracted recent attention is the role that crystal  
86 imperfections of various types can play in changing diffusion behavior (Djimbi et al., 2015;  
87 Gerin et al., 2017; Zeitler et al., 2017b; Fayon and Hansen, 2018), adding to the impact  
88 that imperfections associated with radiation damage have on diffusion kinetics. This focus

89 is not a new concept, as Farley (2000) argued that “*Regardless of model, a critical*  
90 *question for apatite helium thermochronometry is whether the total abundance of defects*  
91 *affects the helium retentivity in the low temperature regime and, if so, how and when the*  
92 *defects are acquired.*”

93 Here, we use a recently developed analytical approach, continuous ramped heating  
94 (hereafter, CRH, Idleman et al., 2018) which is described in **section 2.3**, to assess this  
95 long-standing problem of excess AHe age dispersion by closely examining samples from  
96 a well-characterized geological setting. In a broad survey of natural samples, McDannell  
97 et al. (2018) suggested that CRH should be able to identify variable  ${}^4\text{He}$  outgassing  
98 behavior in the form of differing gas-release components. Our work aims to test this  
99 suggestion using a classic sample suite (Fitzgerald et al., 2006) from the Ferrar Glacier  
100 area of southern Victoria Land in the Transantarctic Mountains that yielded highly  
101 dispersed AHe ages (section 2.1). By better documenting how CRH can reveal  
102 relationships between AHe ages and  ${}^4\text{He}$  outgassing behavior such that it can be  
103 deployed as a routine sample-characterization tool to extract interpretable data from  
104 complex sample sets, we seek to expand our understanding of the nature of He diffusion  
105 systematics in apatite.

## 106 **2. Study Material and Analytical Methods**

### 107 **2.1. The Transantarctic Mountain apatite suite**

108 Ideal apatite samples for our study should (1) have significant dispersed AHe ages, which  
109 is not uncommon, but also (2) be constrained by other thermochronological and  
110 geological information to allow assessment and interpretation of the dispersed ages. The

111 Transantarctic Mountains are a good locality to test our questions because their tectonic  
112 and thermochronological setting is relatively well established and their overall history of  
113 slow cooling since the Mesozoic will tend to amplify any dispersion in AHe ages which  
114 may be due to variations in  ${}^4\text{He}$  diffusion systematics. We have selected a vertical profile  
115 collected from basement granitoids in the Ferrar Glacier area of southern Victoria Land  
116 in the Transantarctic Mountains – Cathedral Rocks – because of its thermal history, as  
117 well as the availability of both AHe and apatite fission track (hereafter, AFT) age  
118 constraints from previous studies (e.g., Fitzgerald 1992, 2002). Those studies suggest  
119 the locality also experienced relatively rapid cooling early in the Oligocene, possibly  
120 resulting from the onset of glacial incision or a change in tectonics.

121 The Transantarctic Mountains (TAM; **Fig. 1**) have long been regarded as an intriguing  
122 feature owing to their large size (>2500 km long), high elevations (>4 km), and the way  
123 they define the western flank of the West Antarctic rift system, in essence separating the  
124 significantly different geological terranes of East and West Antarctica (e.g., Dalziel, 1992;  
125 Fitzgerald, 2002; Goodge, 2020). The West Antarctic rift system underwent two phases  
126 of extension, early initiation in the middle Mesozoic (e.g., Elliot and Fleming, 2004), and  
127 then a later post-Eocene phase (e.g., Wilson et al., 1998; Florindo et al., 2001; Smellie,  
128 2001). In southern Victoria Land basement rock is dominated by the arc-related Cambro-  
129 Ordovician magmatic suite of the Granite Harbour Intrusives (e.g., Allibone et al., 1993)  
130 intruded into polydeformed metasedimentary rocks (e.g., Goodge, 2020) during the Ross  
131 Orogeny. Devonian to Triassic flat-lying sedimentary rocks known as the Beacon  
132 Supergroup were then deposited unconformably on a basement erosion surface. Basin  
133 sedimentation was subsequently ended by extensive basaltic flood magmatism marking

134 the breakup of Gondwana, expressed in southern Victoria Land as the Ferrar Dolerite,  
135 presenting as thick (~300 m) sills within the basement and along the unconformity, as  
136 well as thinner sills distributed within Beacon sediments (e.g., Gunn and Warren, 1962).  
137 Subsequent to Ferrar magmatism, the TAM was formed largely by uplift along the West  
138 Antarctic rift flank (e.g., Fitzgerald, 1992). Due to this rift-flank uplift, the layer-cake  
139 stratigraphy of the TAM dips (1-2°) gently inland before disappearing under the East  
140 Antarctica Ice Sheet (e.g., Gunn and Warren 1962; Goode, 2020).

141 There is a rich collection of thermochronological studies in southern Victoria Land, both  
142 onshore (Gleadow et al., 1984; Gleadow and Fitzgerald, 1987; Fitzgerald and Gleadow,  
143 1988; Fitzgerald 1992, 2002; Olivetti et al., 2018) and offshore (Fitzgerald 2001; Olivetti  
144 et al., 2013) that generally document episodic exhumation with periods of enhanced  
145 cooling and exhumation (though slow relative to most active orogens) in the Cretaceous  
146 and Cenozoic. Fitzgerald et al. (2006) sought to explore the younger part of the  
147 exhumation history (less than ca. 50 Ma) by integrating AFT data with inverse thermal  
148 models, combined with, what at that time, was the relatively new approach of apatite (U-  
149 Th)/He dating. However, apatite (U-Th)/He ages from two vertical profiles collected on  
150 either side of the Ferrar Glacier displayed considerable single-grain age variation (**Fig. 1**).  
151 Thus, the focus of that study shifted from constraining the younger exhumation history of  
152 the TAM towards documenting and exploring why over-dispersion occurs and how such  
153 data might be interpreted. Nevertheless, constraints on the cooling and exhumation  
154 history of this part of the TAM were obtained. With less AHe age dispersion within data  
155 from the north side vertical profile (Peak 1880) the interpretation was more complete: slow  
156 cooling (exhumation) from Late Cretaceous to early Eocene (~1°C/Myr), an increase in

157 cooling rate at ca. 43 Ma, then slowing again until another increase in the late Eocene (ca  
158 37–35 Ma). On the south side of the glacier, AHe data from a vertical profile from the  
159 eastern-most of the peaks of the Cathedral Rocks had much greater age dispersion than  
160 the Peak 1880 profile, thus the interpretation relied mainly on AFT data/models and the  
161 AHe ages added very little to our understanding of the younger cooling history. Apatites  
162 from Cathedral Rocks are therefore the focus of our current study. At Cathedral Rocks,  
163 the interpretation of Fitzgerald et al. (2006) was that cooling/exhumation was relatively  
164 slow (~1°C/Myr) from Cretaceous to the early Cenozoic, with slightly faster  
165 cooling/exhumation beginning ca. 50 Ma (~2.8 °C/Myr).

166 For CRH screening and AHe dating, apatite grains were selected from six samples from  
167 the Ferrar Glacier profile at Cathedral Rocks, originally labeled R22641, R22642, R22643,  
168 R22644, R22645, and R22646 from high to low elevation (Fitzgerald et al., 2006). To  
169 simplify communication, in the following discussion we refer to these as R1, R2, R3, R4,  
170 R5, R6, respectively. All apatite grains were picked, examined, and photographed using  
171 a Nikon SMZ800 microscope under plain light at ~95× magnification for optical  
172 characterization to determine shape and size for calculation of alpha-ejection correction  
173 factors and to assess basic grain characteristics such as presence of inclusions,  
174 euhedral-vs-anhedral shape, and grain integrity (see **Research Data – Table A1**).

175 **2.2 Sample Characterization**

176 **2.2.1 Chemistry**

177 To document their overall composition as well as chemical variability, we analyzed a  
178 number of grains from two samples (R1 and R2) by electron microprobe. Complete results

179 are found in the online archive; Table 1 summarizes results by averaging data for all spots  
 180 for all grains. Only Si, Ce, and F show some modest scatter, but Si and Ce are present  
 181 at low concentrations. The grains are all fluorapatite in composition, with an average  
 182 proportion for Fap:Cap:Hap (Piccoli and Candela, 2002) of 0.859 : 0.007 : 0.133 . Values  
 183 of the fission-track annealing parameter *rnr0* (Ketcham et al., 2007) calculated from the  
 184 elemental analyses range from 0.829 to 0.840, signifying near-endmember fluorapatite  
 185 (**Appendix A – Fig. A.1**).

	R1 N=48 spots, 30 grains				R2 N=51 spots, 30 grains			
	Mean	SD	CDL99	MSWD	Mean	SD	CDL99	MSWD
<b>Si</b>	0.126	0.051	0.006	188	0.150	0.056	0.010	200
<b>Y</b>	0.121	0.053	0.030	7.2	0.151	0.057	0.030	8.4
<b>La</b>	<i>0.044</i>	<i>0.031</i>	0.031	4.3	<i>0.035</i>	<i>0.028</i>	<i>0.030</i>	3.6
<b>Ce</b>	0.195	0.060	0.029	15.4	0.204	0.071	0.030	22.1
<b>Mg</b>	<i>0.003</i>	0.003	0.006	1.0	<i>0.008</i>	<i>0.015</i>	<i>0.010</i>	1.7
<b>Ca</b>	39.63	0.159	0.010	3.7	39.47	0.222	0.010	9.3
<b>Sr</b>	<i>0.014</i>	<i>0.007</i>	0.013	1.1	<i>0.014</i>	<i>0.008</i>	<i>0.010</i>	1.6
<b>Na</b>	<i>0.010</i>	<i>0.007</i>	0.009	2.2	<i>0.012</i>	<i>0.009</i>	<i>0.010</i>	2.2
<b>P</b>	18.53	0.139	0.014	2.0	18.44	0.125	0.010	1.8
<b>S</b>	<i>0.001</i>	0.002	0.006	0.8	<i>0.001</i>	<i>0.003</i>	<i>0.010</i>	1.0
<b>Cl</b>	0.059	0.027	0.009	13.0	0.043	0.023	0.010	3.3
<b>F</b>	3.179	0.203	0.032	54.0	3.290	0.197	0.030	48.0
<b>O</b>	38.62	0.15			38.44	0.16		
<b>TOTAL</b>	100.53	0.36			100.25	0.42		

186 **Table 1.** Electron microprobe analyses of apatites from samples R1 and R2. Means,  
 187 standard deviations, and detection limits are in weight percent. CDL99: Concentration  
 188 with 99% confidence-level detection limit. MSWD (mean square of weighted deviates)  
 189 serves as a measure of scattering of values relative to instrument uncertainties. Values  
 190 in italics are near or below detection limit.

191 **2.2.2 Survey of crystallographic defects**

192 We examined polished sections of grains from samples R1 and R2 to assess the  
193 prevalence of defects in TAM apatites. Before polishing, grains were annealed at 500 °C  
194 for 60 minutes to remove any fission tracks and then etched using two different solutions,  
195 a typical 5M HNO<sub>3</sub> solution commonly used for fission-track etching, and also a 0.5 M  
196 HNO<sub>3</sub> solution to focus on smaller more delicate structures. Details about the size  
197 distribution of defects are part of an ongoing study, but for this paper a key observation  
198 is that TAM apatite grains are highly variable in etchable defect density, ranging from  
199 nearly imperfection-free to being riddled with imperfections of various types (**Fig. 2**).

## 200 **2.3 Data collection and analysis**

201 Individual grains were placed in closed niobium tubes, degassed of their <sup>4</sup>He via the CRH  
202 method at the Lehigh University noble-gas geochronology lab (see below and **Appendix**  
203 **B** for details), and measured for parent U-Th-Sm isotopes via dissolution and isotope  
204 dilution at the Arizona Radiogenic Dating Laboratory with detailed procedures reported  
205 by Reiners and Nicolescu (2006).

### 206 **2.3.1 Continuous ramped heating**

207 The CRH method characterizes the diffusive loss of <sup>4</sup>He by continuous heating following  
208 a progressively increasing temperature schedule, typically at a fixed rate. Evolved He is  
209 measured continuously as a function of time and temperature (Idleman et al., 2018). Our  
210 early experiments (Idleman et al., 2018; McDannell et al., 2018) used a resistance furnace  
211 for heating, which we have now replaced with a fiber-coupled diode laser system. The  
212 laser provides more precise time and temperature control, less temperature lag (i.e.,  
213 better response time), and lower loads of potentially interfering active gases coevolved

214 with He. Here we briefly outline the most recent analytical procedure of our  
215 implementation of CRH and include complete documentation of this CRH procedure in  
216 **Appendix B**. We also provide our observations made on the behavior of standard  
217 Durango apatite to document the behavior of simple  ${}^4\text{He}$  diffusion systematics as  
218 measured by our newest application of the CRH method.

219 **2.3.2 Sample handling**

220 After being selected and photographed, each apatite grain was placed in a closed niobium  
221 (Nb) tube, which had been cleaned and then degassed for 3 hours in a vacuum furnace  
222 at 600 °C. The loaded tube was placed in a hand-made Nb foil envelope ~4 mm in  
223 diameter that had also been prewashed and degassed. We used these small envelopes  
224 to present an even, flat surface to the laser beam in order to achieve better temperature  
225 control and measurement. The packages were placed in quartz-glass holders located in  
226 a mobile sample rack that allows us to load multiple samples and analyze them without  
227 breaking vacuum.

228 **2.3.3 Data collection**

229 Each CRH run was performed under static vacuum conditions with the mass  
230 spectrometer directly open to the sample cell. At the beginning of an analysis, the  
231 extraction line was isolated from its pumping system and the  ${}^4\text{He}$  beam was measured  
232 and recorded for 3 – 4 minutes, allowing us to estimate cold-blank accumulation rates  
233 before the initiation of heating. After heating began, temperatures were recorded by an  
234 optical pyrometer capable of measurement over a range of ~180 to >1200°C. Peaks at  
235 masses 1, 2, 3, 4, and 28 were measured and recorded continuously using a Balzer

236 Prisma Plus quadrupole mass spectrometer. Besides  ${}^4\text{He}$ , the peaks corresponding to H,  
237  $\text{H}_2$ , HD, and  $\text{N}_2$  were monitored because we have found that in some runs high levels of  
238 these active gases can have a moderate impact on  ${}^4\text{He}$  sensitivity by attenuating the  
239 mass 4 beam at temperatures greater than 850 to 900 °C. Two SAES GP50 getter pumps  
240 (operated at 20°C and 300°C) were used to reduce the partial pressures of these active  
241 gases during analysis so that their effects on  ${}^4\text{He}$  were never more than a few percent at  
242 high temperatures when hydrogen and nitrogen attain their highest partial pressures (up  
243 to 100x those seen in the cold background signal).

244 All samples were heated to a temperature of at least 800°C. Samples that continued to  
245 outgas  ${}^4\text{He}$  at 800°C were heated further until they showed no additional  ${}^4\text{He}$  contribution  
246 for at least 1 minute, or until they reached 1100°C (whichever came first). After allowing  
247 2-3 minutes for sample cool-down and additional purification of the evolved  ${}^4\text{He}$ , a  
248 metered aliquot of  ${}^4\text{He}$  of  $\sim 2.22 \times 10^{-13}$  mol was introduced from a pipette system to allow  
249 determination of the total  ${}^4\text{He}$  by the method of standard additions. In some cases, this  
250 post-run cleanup step was preceded by a small increase in the total-release  ${}^4\text{He}$  signal  
251 (rarely exceeding 5%), reflecting gettering of the active gas species suppressing  ${}^4\text{He}$   
252 sensitivity. This small suppression does not have a significant impact on relative patterns  
253 of CRH release behavior but is clearly important to eliminate before measuring the final  
254  ${}^4\text{He}$  abundance for accurate age determination.

255 **2.3.4 Data reduction**

256 During our CRH runs for the Cathedral Rocks apatites, individual crystals were heated at  
257 a fixed rate of 30°C/minute, and temperature and  ${}^4\text{He}$  measurements were recorded

258 every 10 seconds. To smooth noise in the measured sample temperatures, particularly  
259 at low temperatures, we performed a rolling 11-point linear regression of the measured  
260 temperatures and registered the times of  ${}^4\text{He}$  measurement blocks within the regressed  
261 temperature record through interpolation. In practice, sample temperatures determined in  
262 this way agree with the targeted setpoint temperatures defined by the heating schedule  
263 to within 2-3°C. The  ${}^4\text{He}$  beam values were then corrected for dynamic background and  
264 evolved blank, yielding final CRH results in the form of tables of time, temperature, and  
265 corrected  ${}^4\text{He}$  beam values. From these results we calculated the first derivative of  
266 fractional loss ( $f$ ) to construct incremental  ${}^4\text{He}$  loss curves ( $df/dT$  vs  $T$ , hereafter  $df$  as  
267 shorthand) and to extract kinetic data ( $\ln(D/a_2)$ , (1/s), and  $10000/T$  (K)) for each sample.  
268 These data are available in the data repository as **Table A2** and **Table A3**.

269 **2.3.5 Expected behavior: Durango apatite**

270 We carried out CRH analyses of Durango apatite, an apatite standard widely used by the  
271 thermochronology community known for its reproducibility in AHe age (McDowell et al.,  
272 2005) and  ${}^4\text{He}$  diffusion kinetics (Farley, 2000) with two goals: using its degassing  
273 behavior as a benchmark for expected CRH gas-release patterns and using its kinetics  
274 to test analytical reproducibility. We performed CRH screening on grains that were either  
275 internal fragments or abraded spherical balls that were made from air abrasion of  
276 fragments using an apparatus similar to that described by Krogh (1982).

277 The spherical-equivalent radii (Ketcham et al., 2011) of our grains ranged from 99 to 118  
278  $\mu\text{m}$  for the shards and from 90 to 167  $\mu\text{m}$  for the balls. Grains were chosen to provide  
279 variation in size and a fairly large  ${}^4\text{He}$  signal to be measured. In general,  $df$  curves for

280 both the shards and balls (**Fig. 3A**) exhibit the simple and consistent unimodal peaks  
281 predicted by volume diffusion theory (see modeled  ${}^4\text{He}$  outgassing behavior via volume  
282 diffusion under CRH in Idleman et al. (2018) and McDannell et al. (2018)). We  
283 intentionally include results from balls with varying radius to show the precision of our  
284 CRH temperature control. With the same heating schedule of 30°C/min, the larger grains  
285 show a slightly higher-temperature peak-gas release (McDannell et al., 2018) compared  
286 to medium-sized balls (**Fig. 3A**), and the peak-gas release occurred at lower  
287 temperatures for one of the smallest balls despite some moderate roughness.

288 This size-controlled kinetic variability is also evident on the Arrhenius plot (**Fig. 3B**) where  
289 these grains overall show similar behavior but with offsets from published kinetics by  
290 different extents that generally reflect their sizes. After recasting all the results to the same  
291 radius (80  $\mu\text{m}$ , **Fig. 3C**), we effectively removed the effect of grain size. Because the  
292 estimated spherical-equivalent radii of the balls are far more accurate and consistent than  
293 that of the shards, we use the remaining kinetic variation within these balls as an estimate  
294 of temperature uncertainty. At observed values of  $\ln(D/a^2)$  of -14 and -12 1/s, the  
295 calculated temperature ranges for the size-adjusted data are  $311 \pm 8$  °C and  $375 \pm 8$  °C,  
296 respectively, consistent with the variation we observe when we perform calibration of the  
297 optical pyrometer against a reference thermocouple.

## 298 **2.3 Technical issues**

299 Using affordable hardware, a single CRH analysis can be done quickly in about the same  
300 time as a conventional He analysis, and so represents a potentially routine screening tool  
301 that every thermochronology laboratory can perform. There are a few technical issues

302 that are important to appreciate in order to fully evaluate the data from this study. These  
303 technical concerns have implications for measurement of low-temperature kinetics (**Fig.**  
304 **3B, C; Fig. 8**) and overall data precision.

305 The major challenge when undertaking CRH experiments is temperature measurement,  
306 especially in the low-temperature regime (i.e., < 250°C). To increase sample throughput,  
307 we measure temperature for each grain using an optical pyrometer rather than a  
308 thermocouple. The pyrometer was calibrated with a thermocouple-instrumented sample  
309 packet each day before CRH runs. However, two difficulties prevented us from obtaining  
310 highly accurate temperature measurements below ~300°C. First, at the time of sample  
311 analysis, our pyrometer had a lower measurement limit of ~198°C, which didn't allow us  
312 to begin our experiments at temperatures where the very first measurable He release  
313 occurs. Second, due to the time needed for a sample package to reach thermal  
314 equilibrium at low temperatures, we often observed (1) temperatures recorded from the  
315 pyrometer that were lower than that from the reference thermocouple, from right after  
316 laser startup until ~300 °C, both of which were lower than the scheduled linear heating  
317 ramp that was used for data reduction and (2) brief periods of higher-than-expected  
318 temperature readings (laser overshooting) during this time period. The apparent net effect  
319 of these two issues seems mostly to be lower-than-expected  ${}^4\text{He}$  release at low  
320 temperatures that created significant non-linear trends in Arrhenius plots of Durango  
321 apatites (**Fig. 3B, C**), which have been shown to give linear trends during long heating  
322 experiments at low temperatures (Farley, 2000). We call this issue out because even after  
323 later changes to laser software and new hardware (pyrometer) we found the problem  
324 remains, though it is much improved, and so we advise caution in using our current CRH

325 data for fully quantitative measurement of Arrhenius parameters at the lowest  
326 temperatures. Note that this is not a significant problem because in natural samples that  
327 are not from internal shards, both alpha-ejection and diffusion profiles will lead to  
328 concave-upward Arrhenius trends at low temperatures, ruling out use of these low-  
329 temperature data for kinetic quantification in any case. It is worth noting that Farley (2000)  
330 reported lower than expected diffusivities for Durango apatite slabs that had been  
331 polished, so it is also possible that at least for the Durango balls, the lower diffusivities  
332 are the result of near-surface damage due to air abrasion.

### 333 **3. Results from TAM apatite samples**

334 We analyzed 132 single apatite grains from six rocks collected from the Cathedral Rocks  
335 vertical profile (Fitzgerald et al., 2006). For each apatite grain, we obtained its CRH  ${}^4\text{He}$   
336 outgassing curve, AHe age, corrected AHe age (see section 3.4), and  ${}^4\text{He}$  diffusion  
337 kinetics. We also use thermal histories constrained by Fitzgerald et al. (2006) from AFT  
338 data to predict AHe ages using the RDAAM model, allowing us to explore any age  
339 dispersion remaining after removing effects of varying radiation damage and grain size.  
340 These direct results are presented in this section, and raw data for the CRH runs and U-  
341 Th-Sm measurements are included in the data repository (**Table A1, A2, A3**).

#### 342 **3.1. AHe total-gas ages and ${}^4\text{He}$ -outgassing behaviors**

343 For each of the six rock samples, at least 20 grains were analyzed by CRH, and the  
344 single-grain total-gas ages were found to be highly dispersed (**Fig. 4; Fig. 5; Data**  
345 **Repository – Table A1**). The intra-sample age dispersion is not surprising, given the  
346 results obtained by Fitzgerald et al. (2006) and the larger size of our data set. Except for

347 a few old outliers that range up to  $456 \pm 13$  Ma, and one young outlier of  $6.7 \pm 0.2$  Ma,  
348 these apatites have ages ranging from  $27.7 \pm 0.9$  to  $165 \pm 7$  Ma (see **Data Repository** –  
349 **Table A1**). The associated  ${}^4\text{He}$  -outgassing curves (*df*) have a variety of forms, some  
350 similar to that predicted by simple volume diffusion, and others that are quite complex  
351 compared to theoretical behavior. The complex  ${}^4\text{He}$ -outgassing curves are characterized  
352 by sharp gas-release spike(s), delayed gas-release at high temperatures, or frequently a  
353 combination of both. To assist description and discussion we refer to apatite grains having  
354 *df* curves characterized by smooth unimodal peaks as showing “simple” results (i.e., they  
355 passed CRH screening), and samples showing gas spikes and anomalous high-  
356 temperature release as showing “complex” behavior (i.e., they failed CRH screening).

357 All the analyzed apatites show either one or two gas-release peaks, where the earlier  
358 peak always occurred in the range  $572 \pm 45^\circ\text{C}$  (unadjusted for grain size). Less than half  
359 of the apatite grains for each of the six samples survived CRH screening, and these  
360 apatites have greater consistency in gas release, with their peaks occurring at  $590 \pm 35^\circ\text{C}$ .  
361 In a few cases, grains that otherwise passed our criteria for CRH screening have *df* curves  
362 that are broader or narrower than normal and/or show peak gas-release at temperatures  
363 up to  $100^\circ\text{C}$  outside the aforementioned common range (**Fig. 6; Appendix A – Fig. A.2**).  
364 The apatites that survived CRH screening lost at least 90% of their total  ${}^4\text{He}$  between  
365  $\sim 300$  and  $750^\circ\text{C}$ , and their He ages are generally younger, ranging from  $30.6 \pm 1.3$  to  
366  $56.7 \pm 1.0$  Ma. Apatites that failed CRH analysis show moderately or significantly complex  
367  ${}^4\text{He}$  -outgassing behavior and have generally older ages ranging from  $33.4 \pm 1.0$  to  $>100$   
368 Ma. We also found that for each of the six samples, up to seven grains that failed CRH

369 analysis have AHe ages that are older than the AFT central ages reported by Fitzgerald  
370 et al. (2006).

371 **3.2. Effects of radiation damage and grain size**

372 All of our AHe total-gas ages should be influenced by variations in radiation damage and  
373 grain size to some extent. Our TAM apatite suite has a broad range from ~25 to 100 ppm  
374 in effective uranium (hereafter eU;  $[eU] = U + 0.238Th + 0.0012Sm$ ; Cooperdock et al.  
375 (2019)), and  $F_T$  spherical-equivalent radii ranging from ~30 to 75  $\mu\text{m}$ . As in the earlier  
376 AHe single-grain dataset from Cathedral Rocks (Fitzgerald et al., 2006), we found no  
377 obvious relationship between measured total-gas AHe age and eU or grain size for the  
378 entire sample suite (**Fig. 7C, D**). However, potential age-eU and age-size relationships  
379 are evident in the subset of apatites that passed CRH screening (**Fig. 7A, B**).

380 We performed forward modeling by using the thermal histories deduced from AFT data  
381 (Fitzgerald et al., 2006),  $F_T$ -equivalent spherical radius, and measured eU as input for the  
382 HeFTy software (Ketcham, 2005) to predict apparent ages for all of our analyzed apatites  
383 using the RDAAM model. We then normalized our total-gas ages to these RDAAM ages,  
384 calling the resulting ratio the RDAAM-normalization (hereafter, RDN) – samples with  
385 values of 1.0 would have ages predicted from their eU, radius, and reference thermal  
386 history. For each of the six samples, RDNs are still significantly dispersed (**Fig. 4B; Data**  
387 **Repository – Table A1**) with RDN ages ranging from ~0.5 to 3. However, the apatites  
388 that survived CRH screening show a narrower range of RDN (typically ~0.5 to 1.5), and  
389 these ranges are even narrower when considering the RDN values for grains from  
390 individual samples, such as R1, R2, and R6.

391 **3.3. Kinetic variations**

392 Like conventional step-heating analysis, data from CRH analysis allow for the derivation  
393 of kinetic information. Using spherical geometry, cumulative fractional loss, the time  
394 interval between measurements, and the average sample temperature over this interval,  
395 we obtained kinetics data for  ${}^4\text{He}$  diffusion for each grain.

396 We do not use CRH-derived data, at least currently, for precise determination of activation  
397 energy, diffusion coefficient, or closure temperature. Rather, we only explore first-order  
398 intra-sample kinetic variations evident in the data. This is because: (1) compared to step-  
399 heating, CRH's advantage in rapid measurement is offset at very low experimental  
400 temperatures by imprecision in measurement of small gas losses, and (2) temperature  
401 measurements by optical pyrometry are subject to significant systematic offsets below  
402  $\sim 300^\circ\text{C}$  as mentioned previously. Keeping in mind our caution in extracting kinetic  
403 parameters from the lowest-temperature portions of the Arrhenius curves, the overall  
404 locations of these curves in Arrhenius space are sufficiently well defined and precise (see  
405 section 2.2.5) to allow for meaningful comparisons.

406 We present only the kinetics of those apatites that passed CRH screening (Fig. 8),  
407 however Arrhenius plots for all samples are supplied in Appendix A (Fig. A.3). We do this  
408 for two reasons: First, we are particularly interested in exploring intra-sample kinetic  
409 variations between different simply-behaved apatite grains. Second, owing to the fact that  
410 kinetic parameters obtained from both step-heating and CRH are sensitive to fractional  
411 loss of gas, any presence of gas spike(s), which mostly occurs at low to intermediate

412 temperatures, or a second high-temperature release component breaks the linearity and  
413 in fact the justification for Arrhenius relationships.

414 We obtained a wide range of  ${}^4\text{He}$  diffusion kinetics (**Fig. 8A**) for grains giving expected  
415 results, and there is a broad correlation between their ages (total-gas age or RDN age)  
416 and  ${}^4\text{He}$  retentivity, as assessed by relative location on the Arrhenius plot after  
417 normalizing for the effect of grain size (**Fig. 8B**) or normalizing for the collective effect of  
418 grain size and eU (**Fig. 8C**). Among six samples (**Fig. 8C**), R1 and R2 show a clear  
419 correlation between apparent  ${}^4\text{He}$  retentivity and either total-gas age or RDN. Sample R6  
420 also shows such a correlation although it does not show very much intra-sample  
421 dispersion in total-gas or RDN. Kinetic data from the R6 grains show more subtle  
422 variations and less spread on the Arrhenius plot relative to other samples, with the  
423 exception of sample R3. Sample R3 did not show any significant intra-sample variation in  
424  ${}^4\text{He}$  diffusion kinetics, while variations in both its total-gas age and RDN are significant.  
425 Such correlation between age and kinetics is weaker in samples R4 and R5 unless the  
426 oldest age in R4 (array of red points) and the youngest age in R5 (array of blue points)  
427 are not included.

428 **3.4. Age correction**

429 We attempted to correct the ages of those apatites characterized by complex outgassing  
430 behavior by using the peak-fitting process proposed by McDannell et al. (2018). The goal  
431 of performing such age correction is not only to obtain potentially useful age data but to,  
432 more importantly, to explore possible complexities in  ${}^4\text{He}$  diffusion systematics. Based on  
433 the assumption that gas released as spikes and at high temperatures are extraneous with

434 respect to the closure process, we started by making synthetic *df* curves using  
435 established Durango kinetics and spherical geometry in order to fit the first gas-release  
436 peak (i.e., low- to mid-temperature release of gas). This effectively removes gas spikes  
437 and/or delayed gas release at high temperatures (i.e., the second wave of gas release).  
438 We used the first peak because (1) the first peak is almost always located at or close to  
439 the temperatures at which Durango's peak gas release occurs, (2) the second peaks, if  
440 present, often appear at temperatures at or above those at which grains controlled by  
441 Durango kinetics have lost nearly all of their  $^{4}\text{He}$ , and (3) the second peaks occur over a  
442 wide temperature range and often have broad and complex shapes. We discuss the  
443 possible complexity of the delayed gas-release further in **section 4**.

444 Corrected ages were calculated by stripping "extraneous"  $^{4}\text{He}$  from the sample release  
445 using the synthetic *df* curves as a reference, and then applying the measured parent U-  
446 Th-Sm. Obviously, the  $^{4}\text{He}$  correction will always lower ages because the correction  
447 process only removes gas component(s). We found that most of the ages from this apatite  
448 suite correct to younger than ca. 61 Ma, resulting in a much-reduced intra-sample  
449 dispersion (**Fig. 4**; **Fig. 5**). However, we also noted that for each sample some of the  
450 corrected ages are as young as ~20 Ma. This is considerably younger than the youngest  
451 ages obtained from grains that passed CRH screening or other studies from the area and  
452 is probably not plausible geologically (see **section 4.7**).

453 **4. Discussion**

454 Our results from CRH analysis of TAM apatites from the Cathedral Rocks vertical profile  
455 revealed significant intra-sample dispersion in AHe ages, and the dispersion remains

456 even after accounting for effects of grain size and eU. We found that the dispersion was  
457 significantly reduced by CRH screening and that the screened ages broadly correlate with  
458 kinetics. Can these observations be reconciled by a single conceptual model? Below we  
459 relate age dispersion to various types of crystal imperfections, followed by discussion of  
460 gas components, complexities in age correction, and a proposed conceptual model for  
461  ${}^4\text{He}$  retention.

#### 462 **4.1 Radiation damage: only one type of crystal imperfection**

463 Radiation damage, from alpha decay of U and Th, introduces a range of imperfections in  
464 apatite grains that act to slow He diffusion (e.g., Gautheron et al., 2009; Flowers et al.,  
465 2009). Our results suggest that the dispersion observed in the Cathedral Rocks suite  
466 cannot be explained solely by radiation damage, therefore requiring the existence of other  
467 crystal imperfections that augment radiation damage's role in complicating He diffusion.  
468 This interpretation stems from the observation that while samples showing expected  
469 diffusion behavior do exhibit possible correlations of age with grain size and eU, they also  
470 show broad correlations between age and kinetic parameters. Additionally,  
471 crystallographic study of etched apatite grains from this sample suite reveals the  
472 presence of dislocations and sub-grain boundaries that could potentially alter kinetic  
473 parameters (**Fig. 2**).

#### 474 **4.2 Crystal imperfections terminology**

475 To clarify our discussion, we first define some important terms that have had various  
476 usages in the (U-Th)/He literature. First, as used by Farley (2000), the term "defects" or  
477 "damage" refers to a broad range of crystal imperfections stemming from radiogenic and

478 mechanical damage that alters the kinetics of  ${}^4\text{He}$  diffusion. In the more recent (U-Th)/He  
479 literature the term “damage” has been implicitly used as an equivalence for “radiation  
480 damage” because of the development and wide application of the RDAAM model. In order  
481 to avoid miscommunication, we will use crystal “defects” as an overarching term to refer  
482 to finer-scale imperfections and damage resulting from deformation – dislocations, sub-  
483 grain boundaries, grain boundaries, and point defects (Karato, 2008) – as well as point  
484 defects and somewhat larger features associated with radiation damage. Some of these  
485 finer-scale defects (i.e., radiation damage) have been shown to impede diffusion and  
486 have been also termed “traps”. To avoid confusion related to this usage, we prefer to use  
487 the term “sink” to refer to a broader range of probably larger imperfections such as fluid  
488 inclusions and micro-voids that might act as reversible sinks for diffusing He atoms.  
489 Owing to the fact that the term “trap” can depict both objects and processes, in this  
490 document we only use “trap” as a verb to describe processes that temporarily “store” He  
491 atoms in reversible sinks and separately use “radiation damage” when this kind of  
492 diffusion inhibition by defects is mentioned. To summarize, in our usage and discussion  
493 defects are finer-scale imperfections that slow down diffusing He atoms while “sinks” are  
494 larger imperfections that can physically trap He atoms and are possibly reversible.

495 **4.3 Outgassing components**

496 Probably the most obvious feature of gas release from an apatite grain that fails CRH  
497 screening is the delayed release of  ${}^4\text{He}$  at anomalously high temperatures (above  $\sim 700$   
498  $^{\circ}\text{C}$  at a heating rate of  $30^{\circ}\text{C}/\text{min}$  for typical grain sizes), which often represents a  
499 considerable fraction of the total gas and produces a second often unimodal-like gas-  
500 release peak on its *df* plot. This component of gas release might result from any types of

501 crystal imperfections that can act as diffusion sinks. This could include larger sinks such  
502 as fluid inclusions (Baxter, 2003), pores (Lippolt et al., 1994; Watson and Cherniak, 2003;  
503 Domingos et al., 2020), and microvoids (Zeitler et al., 2017b), but could also include  
504 smaller defects like edge dislocations that might also trap helium. These features might  
505 trap diffusing  ${}^4\text{He}$  only temporarily, in proportion to the degree that these sinks are  
506 reversible. Small defects might anneal and return trapped helium to the lattice (Recanati  
507 et al., 2017), but for larger imperfections, the mechanism for this is not clear. Given the  
508 low solubility of He in apatite (on order  $2 \times 10^{-11}$  mol/g-bar; Zeitler et al., 2017b), it would  
509 seem difficult for any trapped He to re-enter the lattice by solution alone since the changes  
510 in pressure that would accompany laboratory heating (2-3x) would be small compared to  
511 solubilities estimated from Henry's Law. It thus seems more likely that an additional  
512 temperature-sensitive mechanism is required to get  ${}^4\text{He}$  in sinks to return to the volume-  
513 diffusion regime in the lattice.

514 Another common feature of the He release from apatites that have failed CRH screening  
515 is sharp spikes of gas release at low to intermediate temperatures. These spikes can at  
516 times account for a considerable fraction of the total  ${}^4\text{He}$  release, though generally being  
517 smaller compared to the broader secondary  ${}^4\text{He}$  released at anomalously high  
518 temperatures. The transient nature of these spikes and their occurrence only at low to  
519 moderate laboratory temperatures suggest that they might be derived from very near-  
520 surface crystal imperfections which have trapped  ${}^4\text{He}$  and then rupture during heating.

521 **4.4 Evaluation of age correction**

522 The results of our attempts at age correction lead us to question the validity of our age-  
523 correction procedure as we applied it, which in itself may provide some critical clues as  
524 to the sources of the various  ${}^4\text{He}$  components recognized in apatites with complex  ${}^4\text{He}$   
525 release. We noted that age correction does not greatly reduce dispersion in these  
526 samples, but it generally shifts ages to lower values that in some cases seem far too  
527 young based on previous thermochronological results from this part of the TAM. The  
528 correction scheme based on the simple removal of  ${}^4\text{He}$  released as spikes and at high  
529 temperatures implicitly relies on the assumption that all of these anomalous components  
530 are “extraneous” (i.e., these components are not part of the syn- and post-closure  
531 radiogenic daughter production) and therefore should be omitted for age calculation.  
532 However, this assumption is likely unfounded for reasons we elaborate below.

533 **4.5 A conceptual  ${}^4\text{He}$  transport model for apatite**

534 Successful models for He accumulation and loss in apatite must be able to reconcile the  
535 observed intra-sample dispersion in AHe ages, anomalous outgassing components, and  
536 kinetic variations. Consider an apatite that acquired a blend of crystal imperfections  
537 including both defects and sinks immediately after its crystallization or perhaps later  
538 during deformation; this would almost certainly be the norm. As this apatite cooled but  
539 was still warmer than its closure temperature for  ${}^4\text{He}$ , diffusing radiogenic  ${}^4\text{He}$  could be  
540 trapped during its random walk in any crystal imperfection it encounters if these  
541 imperfections act as sinks at the current ambient conditions. The accumulation of this  
542 trapped early radiogenic component would be considered to be extraneous with respect  
543 to normal expectations about cooling ages. This trapping would be controlled by the  
544 density of potentially many types of sinks of various sizes. In contrast, at temperatures

545 below the He closure temperature diffusion would be extremely sluggish, and there would  
546 be little or no new trapping of existing or ongoing radiogenic production in sinks. Thus,  
547 natural samples with He sinks would contain radiogenic  ${}^4\text{He}$  in two different locations:  
548 “normal” lattice sites, and sinks, with the sinks filled only with atoms that had undergone  
549 significant numbers of diffusion jumps at higher temperatures. The abundance of  ${}^4\text{He}$  in  
550 sinks would be a function of the number of sinks present, the ability of  ${}^4\text{He}$  to escape from  
551 sinks, and the thermal history, since slow cooling or isothermal thermal histories will  
552 permit extended intervals over which diffusing atoms could encounter a sink, in contrast  
553 to a quenched thermal history in which almost all radiogenic production would occur at  
554 lower temperatures.

555 When outgassing of these sink-bearing apatite crystals is undertaken in the laboratory,  
556  ${}^4\text{He}$  atoms that had not previously encountered a sink would begin to diffuse through the  
557 crystal, and those that did not encounter a sink on the way out would be released to define  
558 the first gas-release peak of the  $df$  curve. However,  ${}^4\text{He}$  atoms that are part of this “normal”  
559 diffusing component would also have a high probability of encountering sinks during  
560 laboratory heating. Thus, trapping could occur during two phases: for higher-temperature  
561  ${}^4\text{He}$  components (i.e., at temperatures above closure) that are produced or incorporated  
562 into the crystal in geologic times, and for all  ${}^4\text{He}$  components during outgassing in the  
563 laboratory. Finally, at higher laboratory temperatures, any  ${}^4\text{He}$  trapped in sinks, including  
564  ${}^4\text{He}$  that was trapped both in natural and in laboratory processes, would see a high  
565 probability of escaping to return to the volume-diffusion regime, and be manifested in the  
566 second high-temperature release peak around  $\sim 800^\circ\text{C}$ . In detail, given the tortuous  
567 random-walk path taken by  ${}^4\text{He}$  atoms as they migrate through apatite crystals, it is

568 conceivable that some  ${}^4\text{He}$  atoms might become trapped and released multiple times,  
569 depending on the kinetics of trap escape. In this model, AHe ages that are greater than  
570 their AFT counterparts would be geologically meaningful.

571 Zeitler et al. (2017b) pointed out that given the large number of diffusion jumps needed  
572 for an atom to escape from a crystal, the probability becomes very high that an atom will  
573 encounter a feature within the lattice. This agrees with our conclusion that gas-release  
574 spikes are sourced from imperfections located very near the grain surface and our  
575 observation that even grains with very complex behavior show only a few such spikes.  
576 Overall the number of sinks within a grain need not be very large or voluminous to  
577 significantly alter diffusion systematics within a grain.

578 In this model, an age correction scheme that removes all of the laboratory high-  
579 temperature gas-release would result in an underestimation of the amount of  ${}^4\text{He}$   
580 expected from closure theory and therefore yield an overcorrected age. The degree of  
581 overcorrection would be worst for a rapid-cooled sample in which almost all trapping  
582 happened in the laboratory, and least for a sample taken directly from the partial retention  
583 zone, for which a significant fraction of its  ${}^4\text{He}$  found in sinks arrived there in nature.

584 What is challenging but interesting is that the ratio of low- to high-temperature gas release  
585 in the laboratory is a sample-specific property that will be controlled not only by the  
586 amount and type(s) of sinks but also by the thermal history. These factors will determine  
587 what proportion of the  ${}^4\text{He}$  content was geologically mobile and thus prone to trapping  
588 under natural conditions. Thus, the presence of a secondary CRH release peak is most

589 directly an indication that sink-related crystal imperfections are present in the sample.  
590 Whether a useful age correction scheme can be developed under this model is unclear  
591 and will require further work. However, it is also worth noting that for a given sample, all  
592 of the grains will have experienced the same thermal history, therefore the remaining  
593 dispersion in AHe ages after accounting for grain size and radiation damage via eU  
594 implies the presence of other features that have influenced the accumulation and release  
595 of He. This model also presents a new possibility in AHe thermochronology in that the  
596 pre- and syn- closure accumulation of extraneous  $^{4}\text{He}$  and its laboratory release at high  
597 temperatures may offer the potential to recover additional constraints on thermal history  
598 (e.g., at earlier times at higher temperatures). Whether this is so will depend on  
599 understanding the kinetics of imperfections in apatite, and so two targets for future  
600 research are understanding the degree to which imperfections of different types and sizes  
601 can anneal or not, and if not, what mechanisms nonetheless return helium to the lattice.  
602 Finally, another interesting unanswered question is whether the presence of larger  
603 imperfections as revealed by CRH analysis would stand as a proxy for the presence of  
604 finer-scale defects that might alter diffusion kinetics but not lead to trapping.

#### 605 **4.6 Kinetic variability beyond radiation-damage models**

606 CRH results from apatites that pass CRH screening shed light on the first-order kinetic  
607 variability of  $^{4}\text{He}$  diffusion. Arrhenius plots from simply-behaved grains show that a  
608 significant range of intra-sample kinetic variability remains after grain size and eU effects  
609 are taken into account (Fig. 8). Additionally, while we observed some modest variations  
610 in the small amounts of Si and Ce present in TAM apatites (Section 2.2.1), these  
611 variations are not likely to lead to significant variations in the kinetics of fission-track

annealing, nor do variations in the Cl and OH content of these fluorapatites (Barbarand et al., 2003). One might infer that this limited compositional control on fission-track annealing could apply to  ${}^4\text{He}$  diffusion kinetics as well. Published results relating apatite composition to changes in  ${}^4\text{He}$  diffusivity are mixed and likely complicated by the use of the FT annealing proxy  $D_{\text{par}}$ . However, there is a direct correlation between apatite composition and track annealing. Djimbi et al. (2015) discussed calculations showing that endmember fluorapatite and chlorapatites *should* have somewhat different He diffusion kinetics, however the small compositional variation observed in the TAM apatites suggests that halogen content is not likely to be a significant source of the kinetic variations we observed.

Past experimental and modeling studies have shown that He diffusivity in apatite is influenced by defects in the form of fine-scale radiation damage (Flowers et al., 2009; Gautheron et al., 2009; Shuster and Farley, 2009), with pristine lattices being associated with much more rapid diffusion. By analogy, we would argue that the defects we observed to be present in varying amounts (**Section 2.2.2**) are altering the samples' diffusion kinetics to different degrees, augmenting diffusivity changes due to radiation damage. This is consistent with the primary data presented by Flowers et al. (2009), in which radiation damage is clearly a first-order control on retentivity but considerable scatter remains that we would argue reflects the presence of other defects, the impacts of which would matter particularly in settings involving slow cooling or thermal stagnation, where greater single-grain age dispersion often appears (e.g., Reiners and Farley, 2001; Fitzgerald et al., 2006). An important area of future research will be finding non-

634 destructive methods to characterize the types and sizes of imperfections present in  
635 analyzed samples.

636 **4.7 Application of CRH screening to TAM exhumation**

637 For the Cathedral Rocks, Fitzgerald et al. (2006) documented that relatively slow cooling  
638 through the late Cretaceous ( $\sim 1$  °C/m.y.) was followed by slightly more rapid cooling ( $<$   
639 3 °C/m.y.) as constrained by the AFT age-elevation profile and inverse thermal-history  
640 modeling. However, as discussed above, their over-dispersed AHe single-grain ages  
641 were not able to further constrain the cooling/exhumation history. A final question that  
642 remains is whether, after extensive analysis, we are able to place better constraints on  
643 the Eocene-Oligocene cooling low-temperature history for the Cathedral Rocks profile.  
644 Our samples that passed CRH screening still show considerable age dispersion, and a  
645 critical question for us is whether and how to assign thermochronologically meaningful  
646 weight to each AHe age.

647 We would argue that without assessment of individual-grain diffusion kinetics, the  
648 significance of thermal histories determined from the individual AHe ages is unclear. In  
649 the case of our samples, the ability of CRH analysis to screen apatites based on  
650 consistent criteria – particularly unimodal  $df$  curves – permits us to focus on a less  
651 dispersed subset of our data. Additional screening allows us to identify a further subset  
652 of grains with  $^4\text{He}$  diffusion kinetics similar to those of Durango apatite, and all our other  
653 grains are considerably more retentive: just using the offset in diffusivity seen across  
654 samples, some might be up to 25 to 30°C more retentive in closure temperature than

655 Durango apatite, meaning that their retentivity (i.e., temperature sensitivity) begins to  
656 approach that of fission tracks in apatite.

657 If we simply focus on ages from CRH-screened grains that have kinetics close to those  
658 of Durango, we do obtain better temporal constraints and more definitive evidence of a  
659 faster-cooling episode in the Cenozoic (**Fig. 9, filled diamonds**). This new interpretation  
660 allows us to infer that more rapid rock exhumation began at or by 35-40 Ma and is quite  
661 consistent with the thermal history proposed for the Peak 1880 profile on the north side  
662 of the Ferrar Glacier (**Fig. 1; Fig. 9**) where AHe dates are less dispersed. This signal has  
663 been interpreted as tectonic, either due to dextral-transstension (Olivetti et al., 2013, 2018)  
664 or rifting and escarpment retreat further south along the Transantarctic Mountain front  
665 (Miller et al., 2010). Such increase in cooling/exhumation rate at ca. 35 Ma is also a  
666 cooling/exhumation signal seen at a number of locations along the TAM, and this  
667 enhanced exhumation in the Late-Eocene-early Oligocene has been alternatively  
668 attributed to enhanced erosion due to the onset of glaciation in Antarctica at ~35 Ma (e.g.,  
669 He et al., 2021; Thomson et al., 2019).

670 **5. Conclusions**

671 Assessment of outgassing components and evaluation of age corrections suggests that  
672  ${}^4\text{He}$  transportation in apatite might be controlled by mechanisms that are more complex  
673 than those considered in current models for the (U-Th)/He system. We argue that CRH  
674 analysis can be used to empirically screen apatites for  ${}^4\text{He}$  components trapped in a  
675 variety of high-retentivity features presented in some apatite crystals (i.e., sinks), and  
676 additionally permit assessment as to whether  ${}^4\text{He}$  diffusion is occurring as expected from

677 existing volume diffusion and radiation-damage models. Our reinvestigation of an age-  
678 dispersed suite of apatites from the Cathedral Rocks of the TAM using CRH analysis  
679 documents the presence of such retentive  ${}^4\text{He}$  sinks in many of the analyzed apatite  
680 grains. Screening of the results has allowed us to identify a subpopulation of the analyses  
681 comprised of the youngest cluster of the AHe ages that exhibit unimodal  ${}^4\text{He}$  outgassing  
682 behavior and Durango-like kinetics that suggest rapid rock exhumation at *ca.* 35 Ma,  
683 consistent with other geological evidence suggesting regional initiation of glaciation at the  
684 end of the Eocene (ref. Ivany et al., 2006).

685 Our findings underscore the important message that simple volume diffusion and current  
686 radiation-damage models may be insufficient to fully describe (U-Th)/He systematics in  
687 some apatite populations. Experiments using CRH can characterize  ${}^4\text{He}$  abundance and  
688 characterize  ${}^4\text{He}$  release behavior in as little as 30 minutes per aliquot, making it suitable  
689 for routine dating. Having information such as this in hand before attempting age  
690 interpretations and modeling seems to us far more preferable than relying on statistical  
691 manipulation or analysis of numerous grains to address age dispersion after the fact. We  
692 recommend more widespread deployment of CRH as well as  ${}^4\text{He}/{}^3\text{He}$  analysis for routine  
693 AHe dating, especially for sample suites that show significant age dispersion.

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709 **Research Data**

710 Research Data associated with this article can be accessed via Harvard Dataverse at  
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914 **Figure Captions**

915 **Figure 1.** (A) Simplified map of southern Victoria Land showing location of the two vertical  
916 sampling profiles from Peak 1880 and Cathedral Rocks. Filled areas are ice-free. TAM:  
917 Transantarctic Mountains. (B and C) Composite AFT and AHe plots summarizing the  
918 preferred cooling/exhumation path (dashed line) for the two vertical profiles but with AHe  
919 data moved down in elevation relative to the AFT data based on the difference in closure  
920 temperatures (30 – 35 °C) divided by the geologically constrained paleogeothermal  
921 gradient (20 – 25 °C/km). Modified from Fitzgerald et al. (2006). See that paper for a  
922 detailed explanation. For the AHe data, Fitzgerald et al. (2006) judged that younger AHe  
923 ages (between the minimum age and a weighted mean) were more likely to constrain a  
924 “true” thermal history, perhaps an early precursor to using screened CRH-screened ages  
925 (as shown in this study, section 4.7). Less dispersion in the Peak 1880 AHe data allowed  
926 constraints to be placed on Late Eocene cooling, whereas much greater dispersion in  
927 AHe data from Cathedral Rocks precluded such constraints.

928 **Figure 2.** Photomicrographs of annealed, polished apatite grains etched in 0.5M HNO<sub>3</sub>,  
929 showing large range of imperfection densities and types. Long straight features are  
930 polishing scratches.

931 **Figure 3.** Incremental <sup>4</sup>He release (*df*) curves (A), and CRH-derived Arrhenius  
932 relationships (B) of Durango apatites under a 30 °C/min heating ramp rate. Gray curves  
933 and circles depict results from Durango shards; blue curves and triangles show results  
934 from Durango balls. Red dashed line marks diffusion kinetics of <sup>4</sup>He in Durango apatite  
935 acquired by Farley (2000) via step-heating, adjusted to a radius of 80 µm. (C) Arrhenius  
936 plot of the same analyses adjusted to a common spherical-equivalent radius of *a* = 80 µm  
937 in order to assess precision in the kinetics obtained by CRH, indirect verification of  
938 temperature control. (D) shows the temperature uncertainties at two points for CRH  
939 analysis of Durango balls.

940 **Figure 4.** Single-grain AHe ages (upper plot) and RDAAM-normalized relative ages  
941 (lower plot). Ages from each sample are presented in individual panels. Each plot includes  
942 measured single-grain total-gas age, screened age, and CRH-corrected age,  
943 accompanied by their kernel density estimations (KDEs). For better visualization ages  
944 older than 100 Ma were omitted, so were the age/RDAAM greater than 3; *n* represents  
945 numbers of plotted data points; Table A1 shows information for all ages. See text for  
946 discussion of different age types show here and in Figure 5.

947 **Figure 5.** Cathedral Rocks AHe and AFT age-elevation plots summarizing results and  
948 interpretations from this study and from Fitzgerald et al. (2006). Light blue circles and  
949 curves show single-grain total-gas AHe ages (this study) and their KDE, respectively;  
950 dark blue circles and shaded curves show single-grain screened AHe ages (this study)  
951 and their KDE; pink shaded curves show KDE of single-grain CRH-corrected ages;  
952 red circles show AHe weighed mean ages (Fitzgerald et al., 2006); orange triangles show  
953 AFT central ages (Fitzgerald et al., 2006); red dashed line shows the best-fit trend to the  
954 weighted mean AHe ages from Fitzgerald et al. (2006).

955 **Figure 6.**  $df$  curves measured using a 30 °C/min ramped heating schedule. Results from  
956 each sample are presented in individual subpanels. (A) Results from grains that passed  
957 CRH screening. (B) Results from grains that failed screening. (C) Results from all grains,  
958 color-coded into three groups: blue, grains whose AHe ages are not older than the oldest  
959 screened age in that sample; red, grains whose AHe ages are older than the fission-track  
960 central ages measured by Fitzgerald et al. (2006); orange, grains whose ages fall  
961 between the other two groups.

962 **Figure 7.** (A) Relationships between single-grain AHe age and grain size (total  
963 compilation shown in panel C). (B) Relationships between single-grain AHe age and  
964 equivalent uranium (eU) (total compilation shown in panel D). Sizes are calculated as  $F_T$   
965 -equivalent spherical radius. To be consistent with prior publications eU is calculated as  
966  $U + 0.235^*Th$ ; Cooperdock et al (2019) provide a more accurate update). Filled circles  
967 show grains that passed CRH screening.

968 **Figure 8.** CRH-derived Arrhenius relationships of apatite grains that passed CRH  
969 screening, showing relationship between AHe ages and apparent  ${}^4He$  diffusivity. (A)  
970 Observed kinetics color-coded by observed age. (B) Kinetics adjusted to 80  $\mu m$  radius,  
971 color-coded by observed age. (C) Kinetics adjusted to 80  $\mu m$  radius, color-coded by  
972 RDAAM-normalized relative age. Dashed green line shows the kinetics of  ${}^4He$  diffusion in  
973 Durango apatites measured by Farley (2000). Vertical dotted line provides visual  
974 reference to aid comparison of diffusivity variations at  $10000/K = 15$  ( $\sim 393.5$  °C).

975 **Figure 9.** Composite AFT and AHe plot (constructed as Figure 1C) summarizing our  
976 updated cooling/exhumation path for the Cathedral Rocks locality. Preferred path for  
977 Peak 1880 is from Fitzgerald et al. (2006).

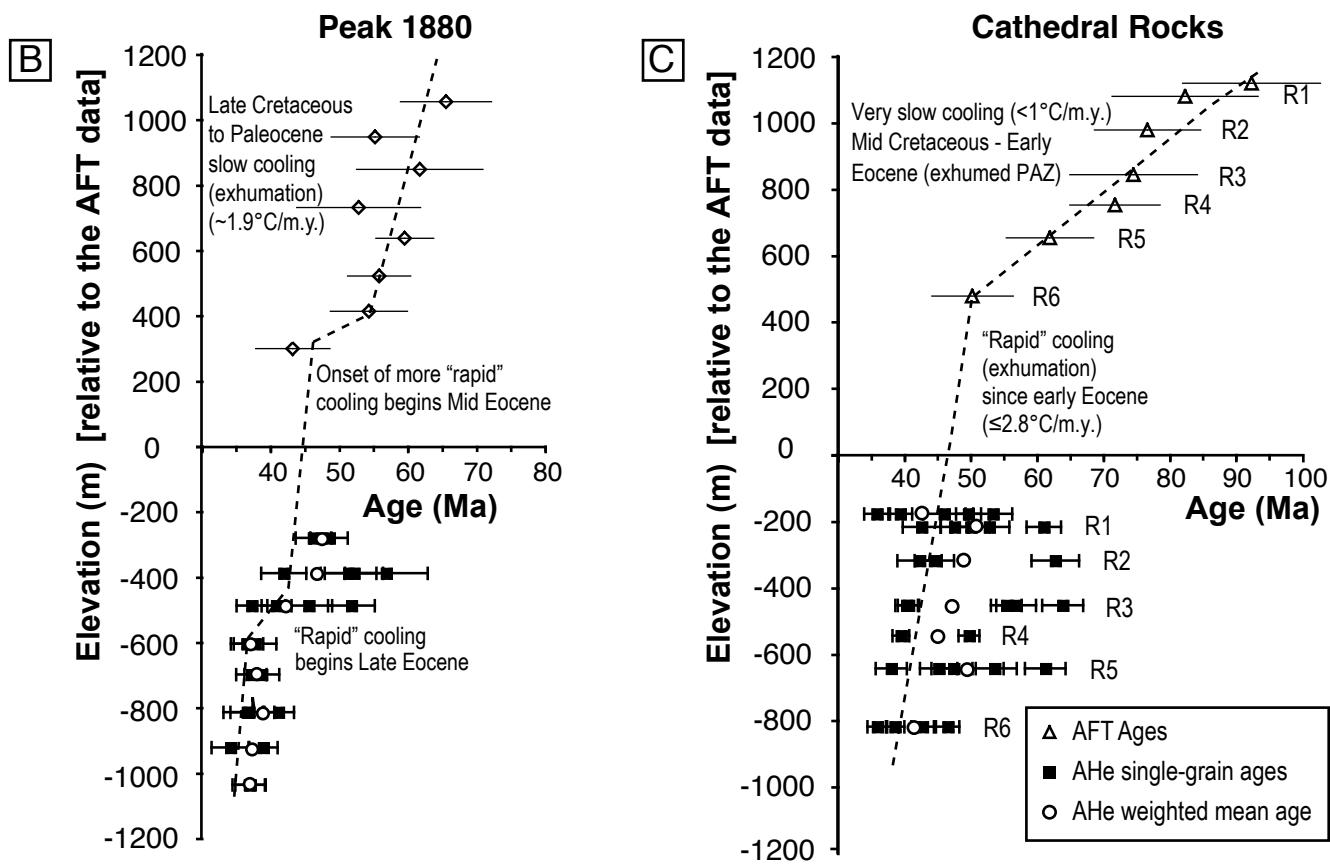
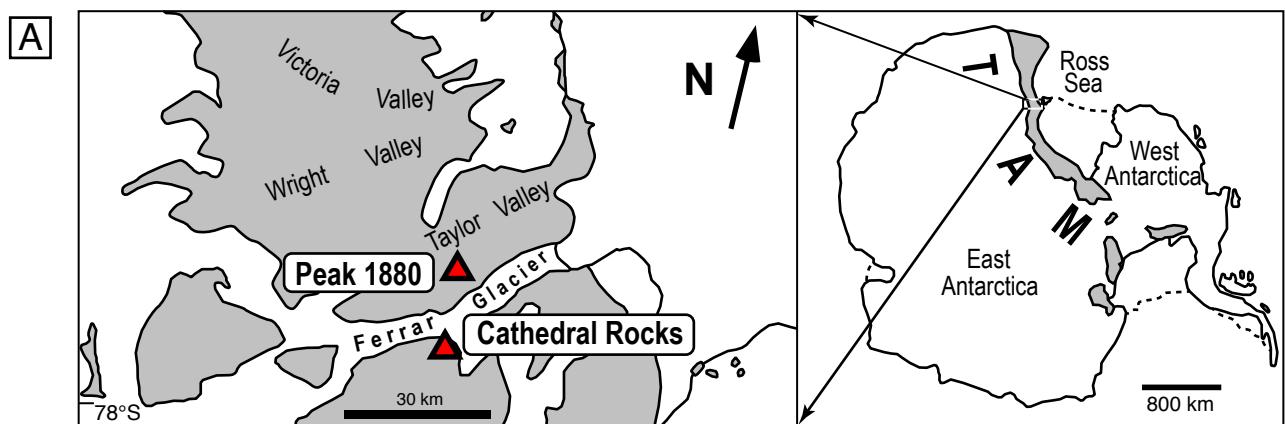


Figure. 1

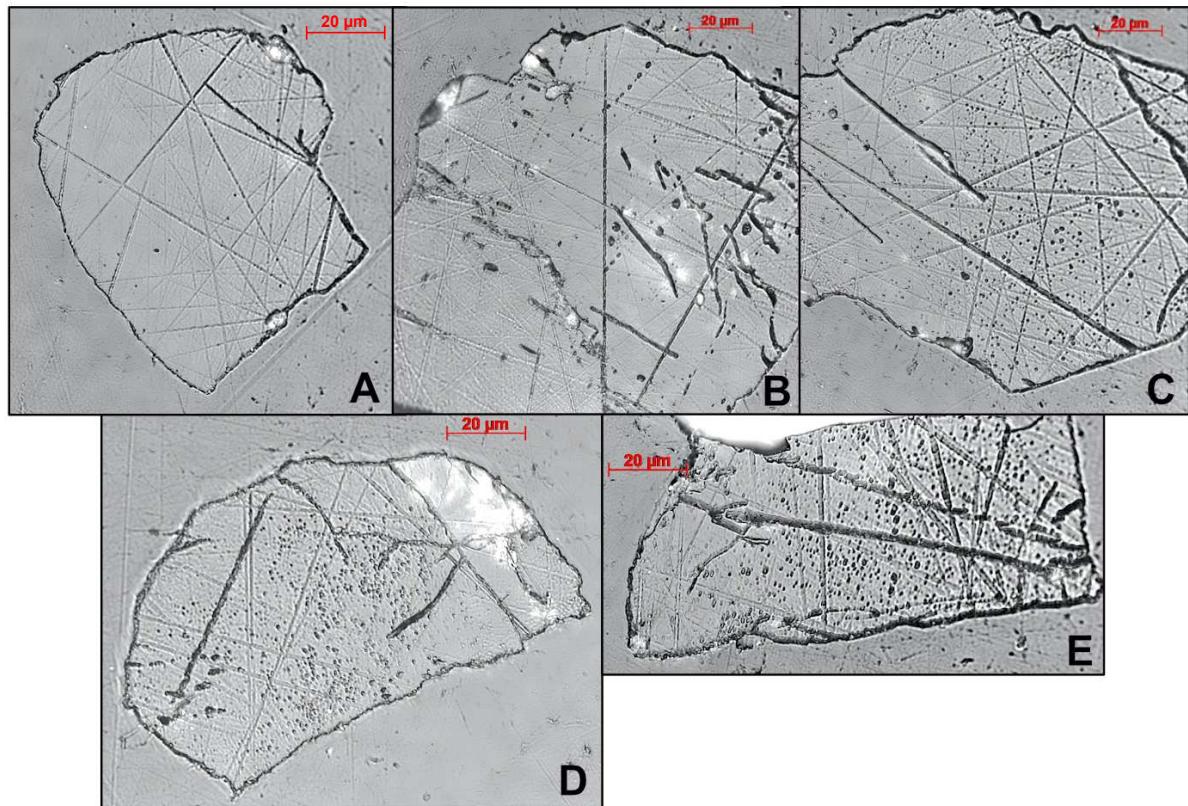


Figure 2

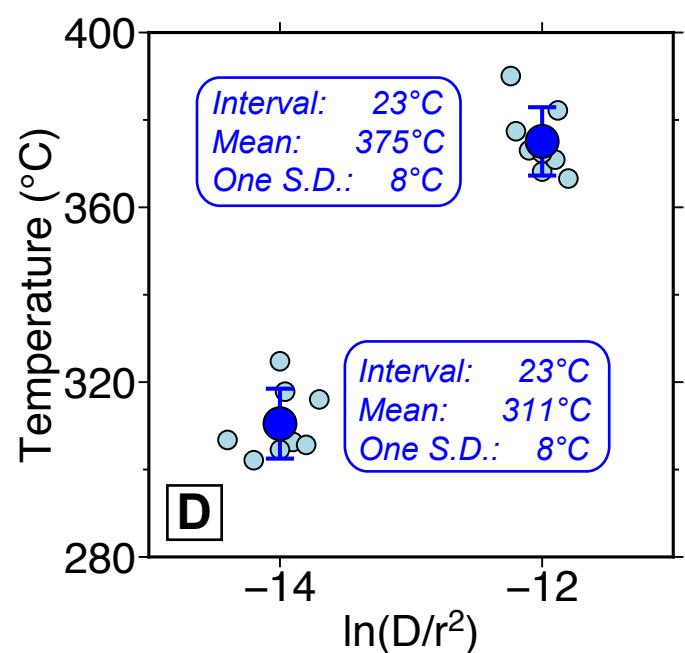
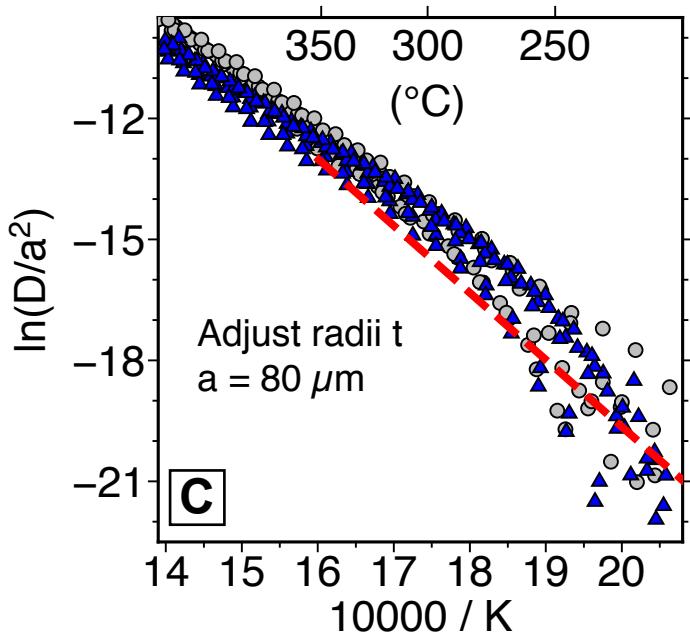
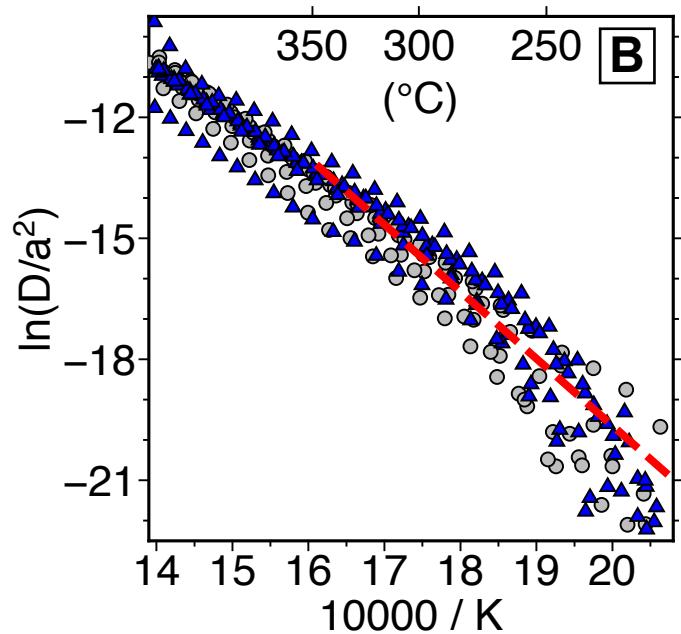
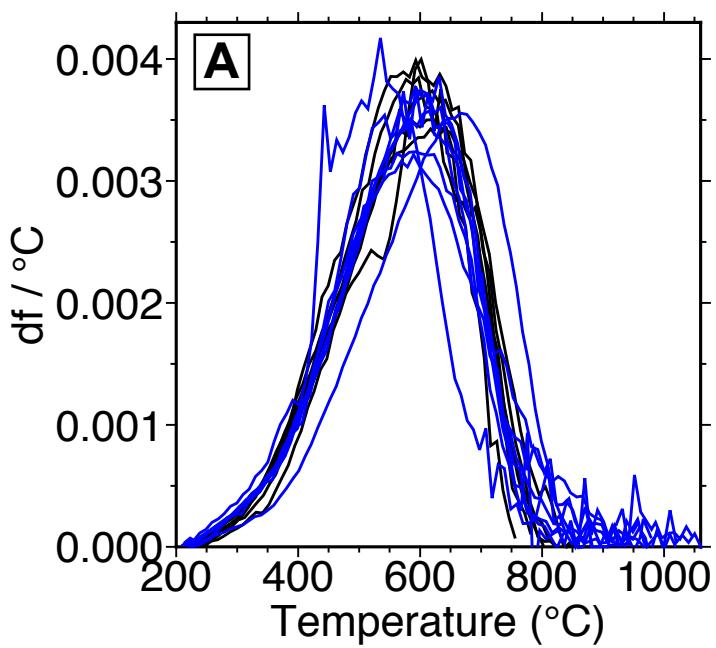


Figure 3

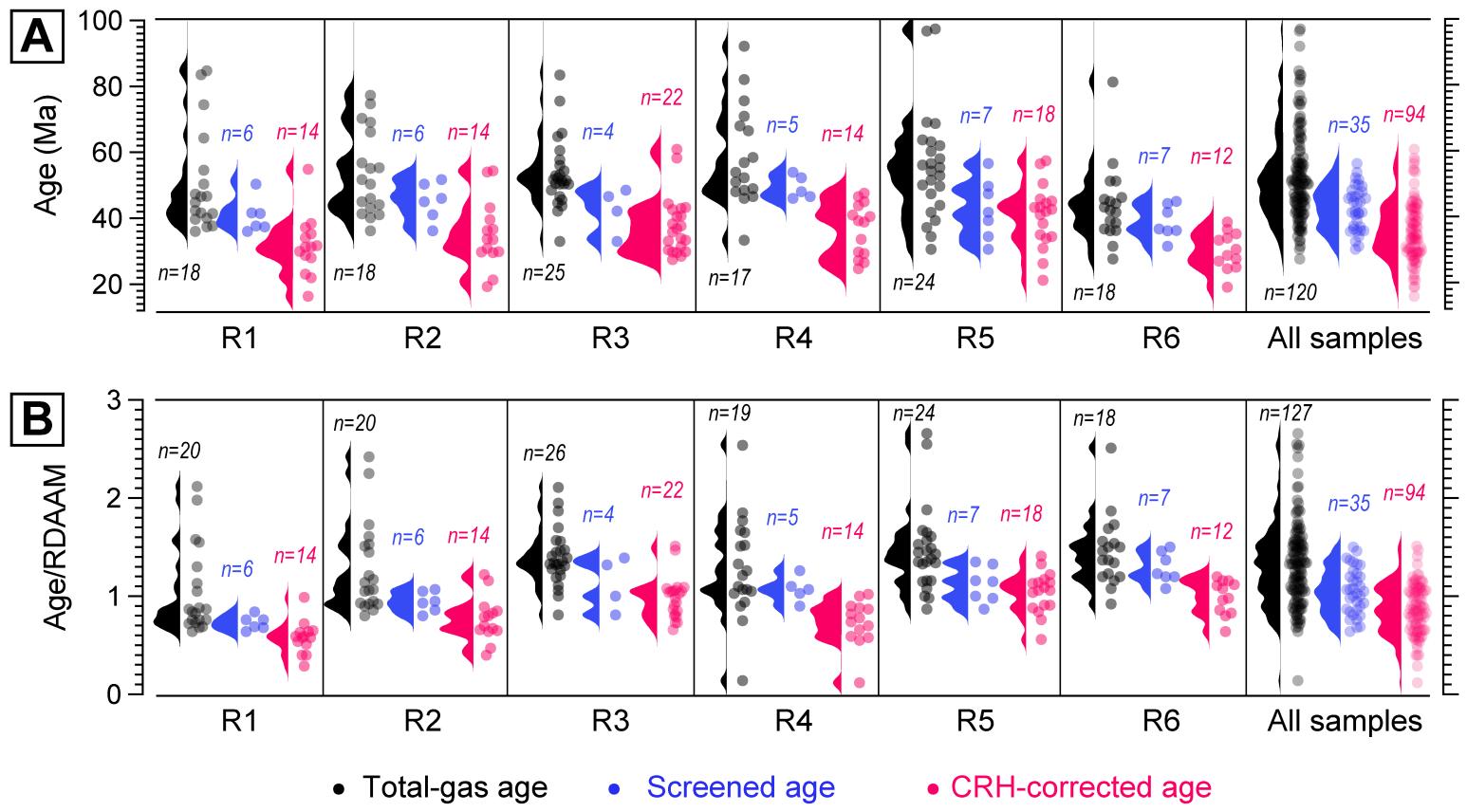


Figure 4

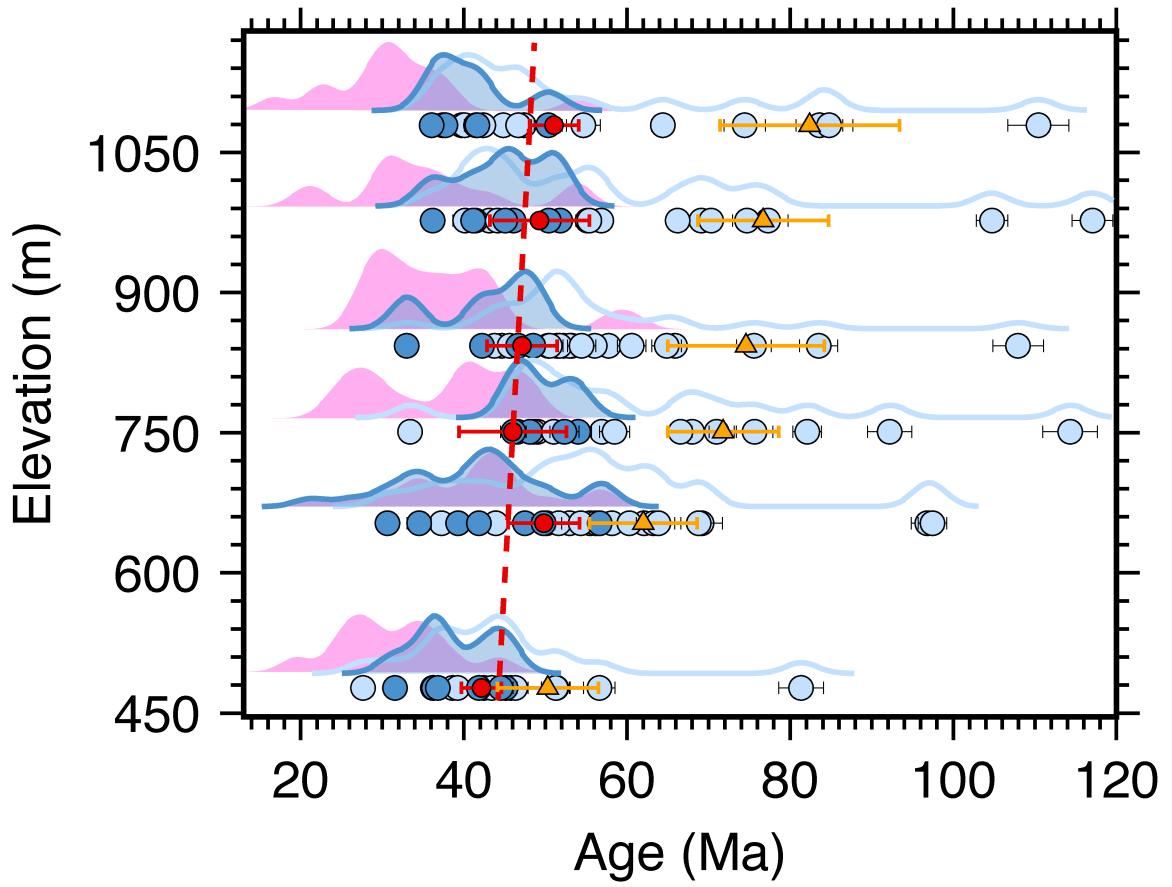


Figure 5

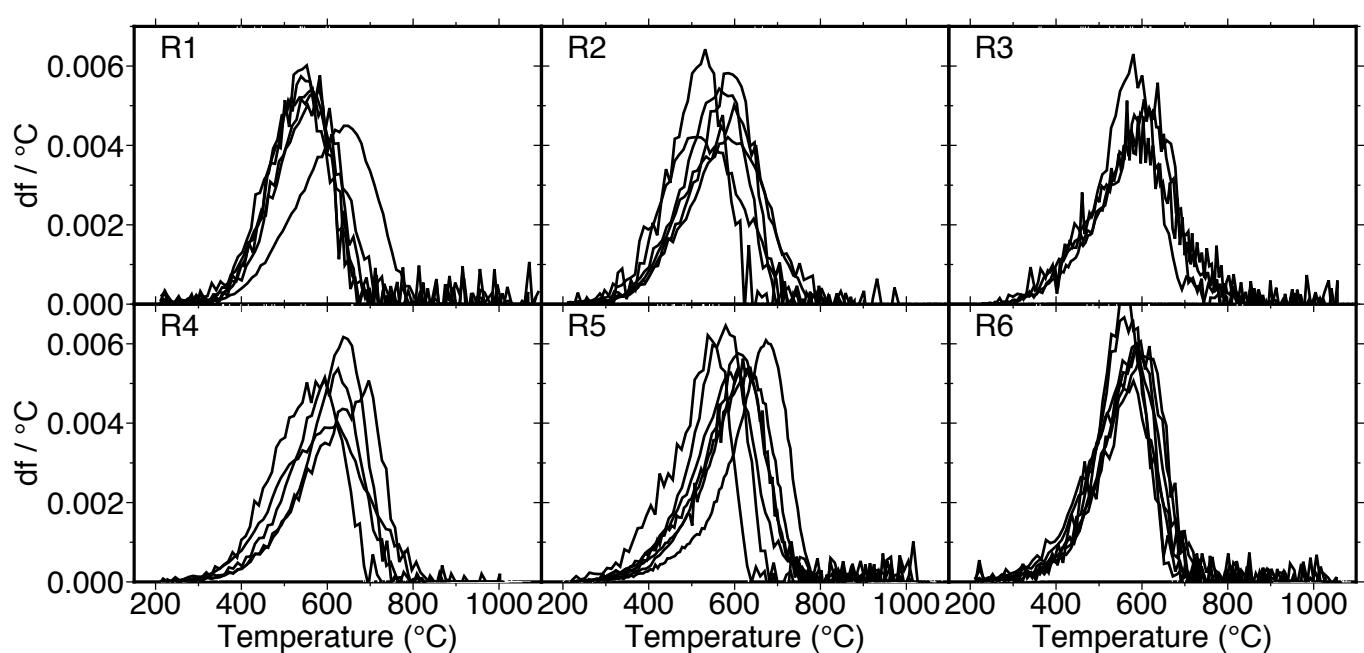
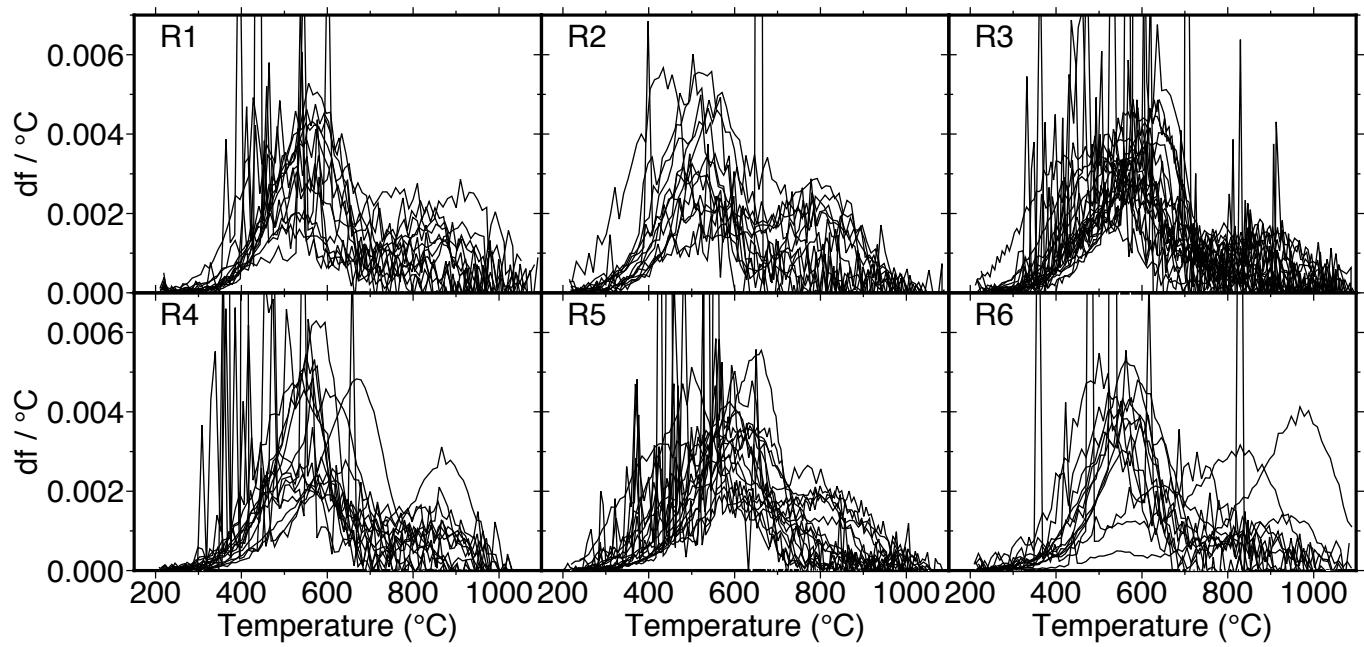
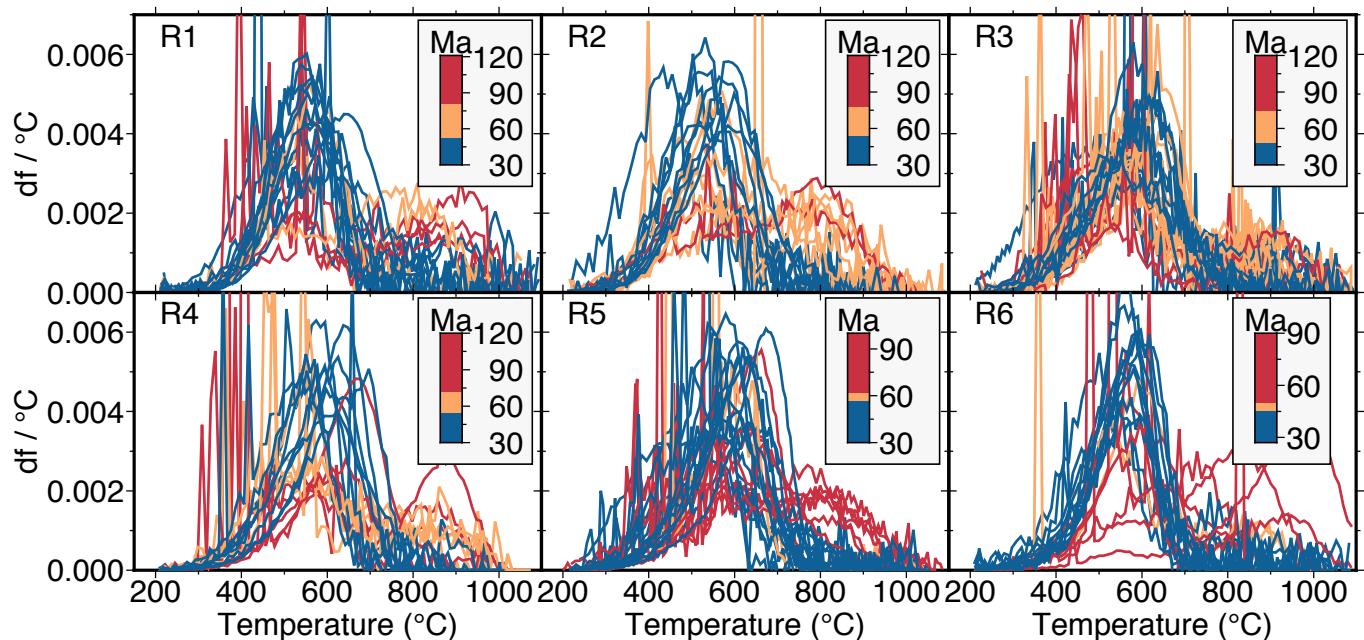
**A****B****C**

Figure 6

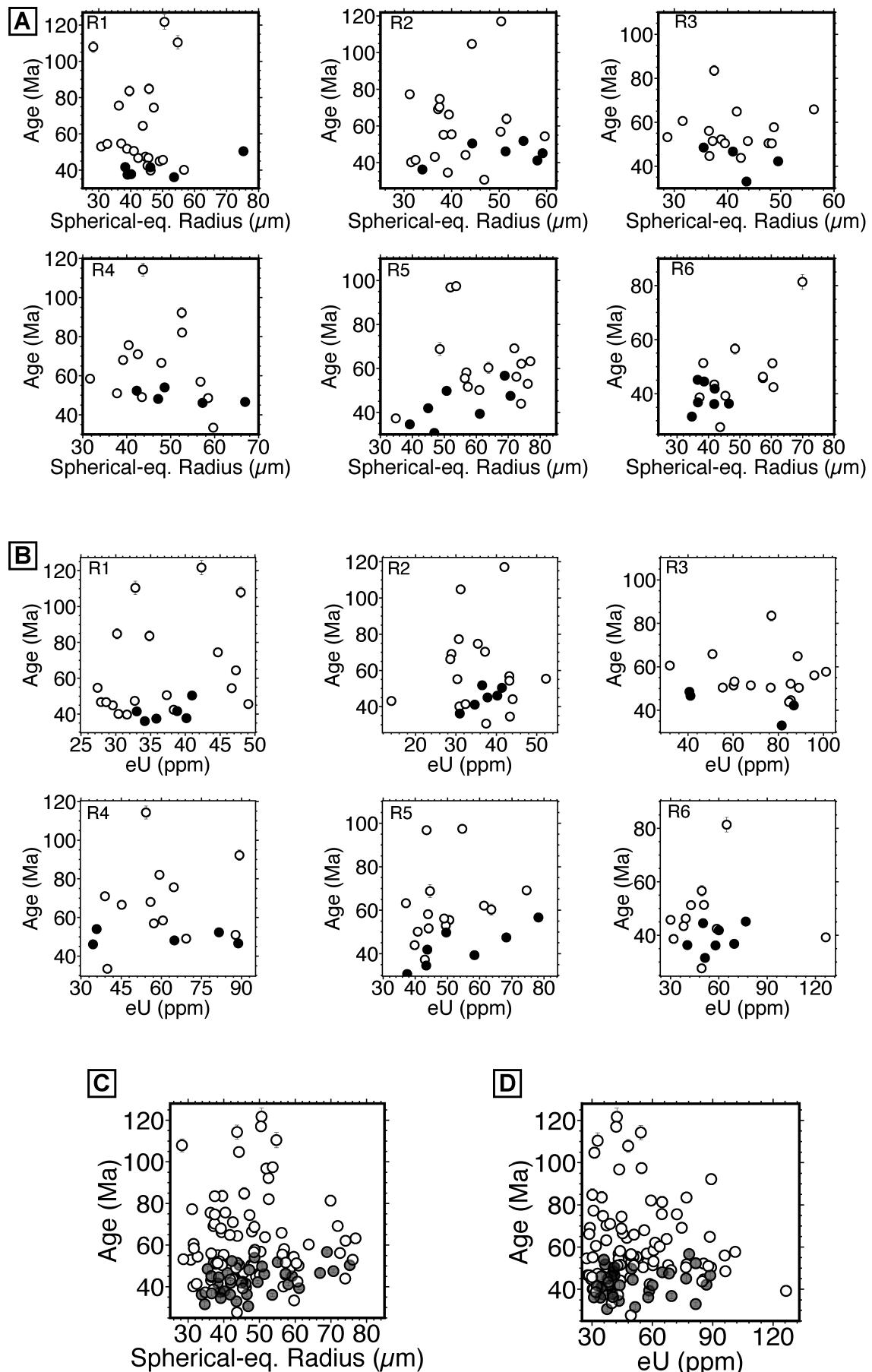


Figure 7

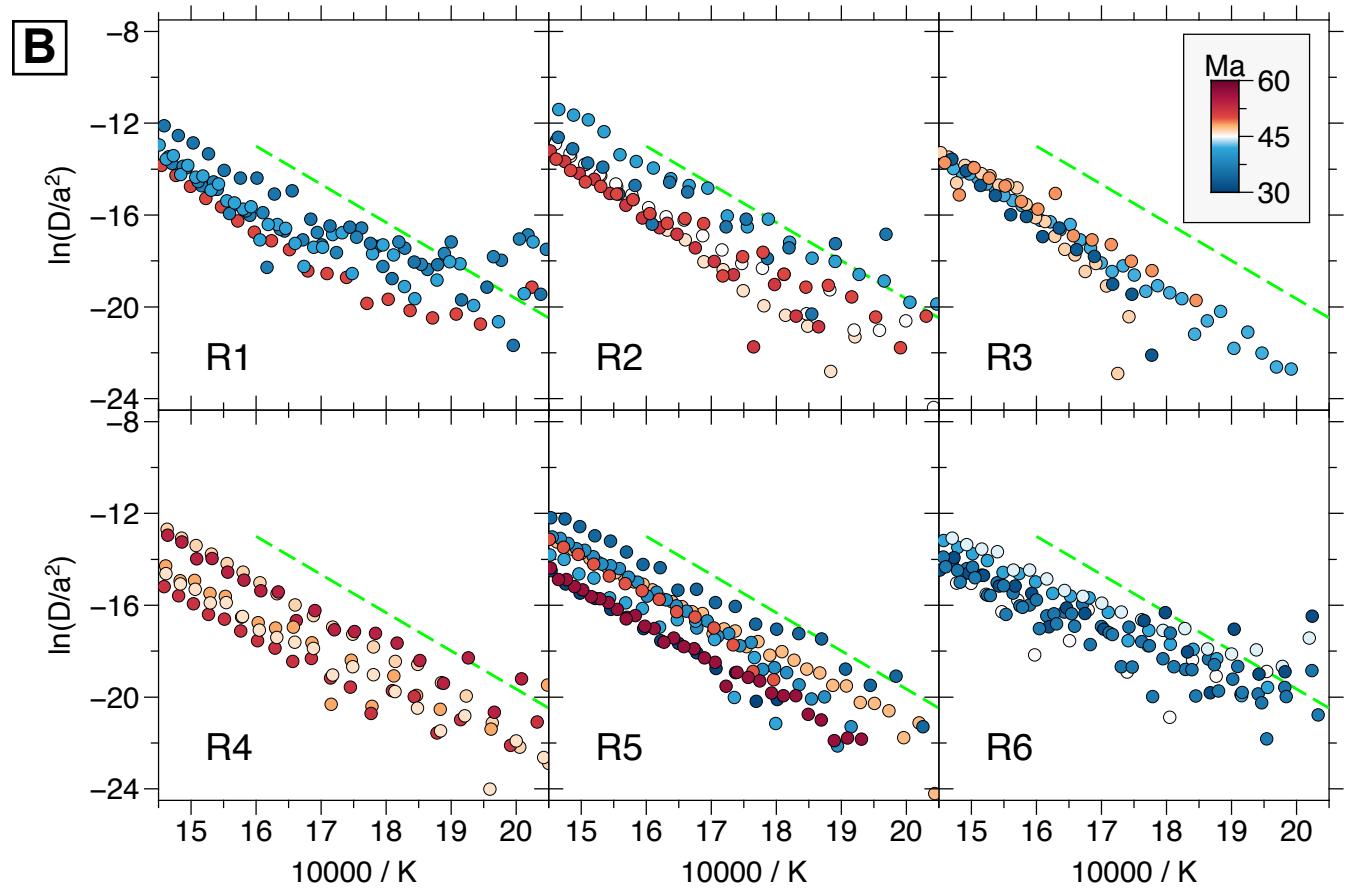
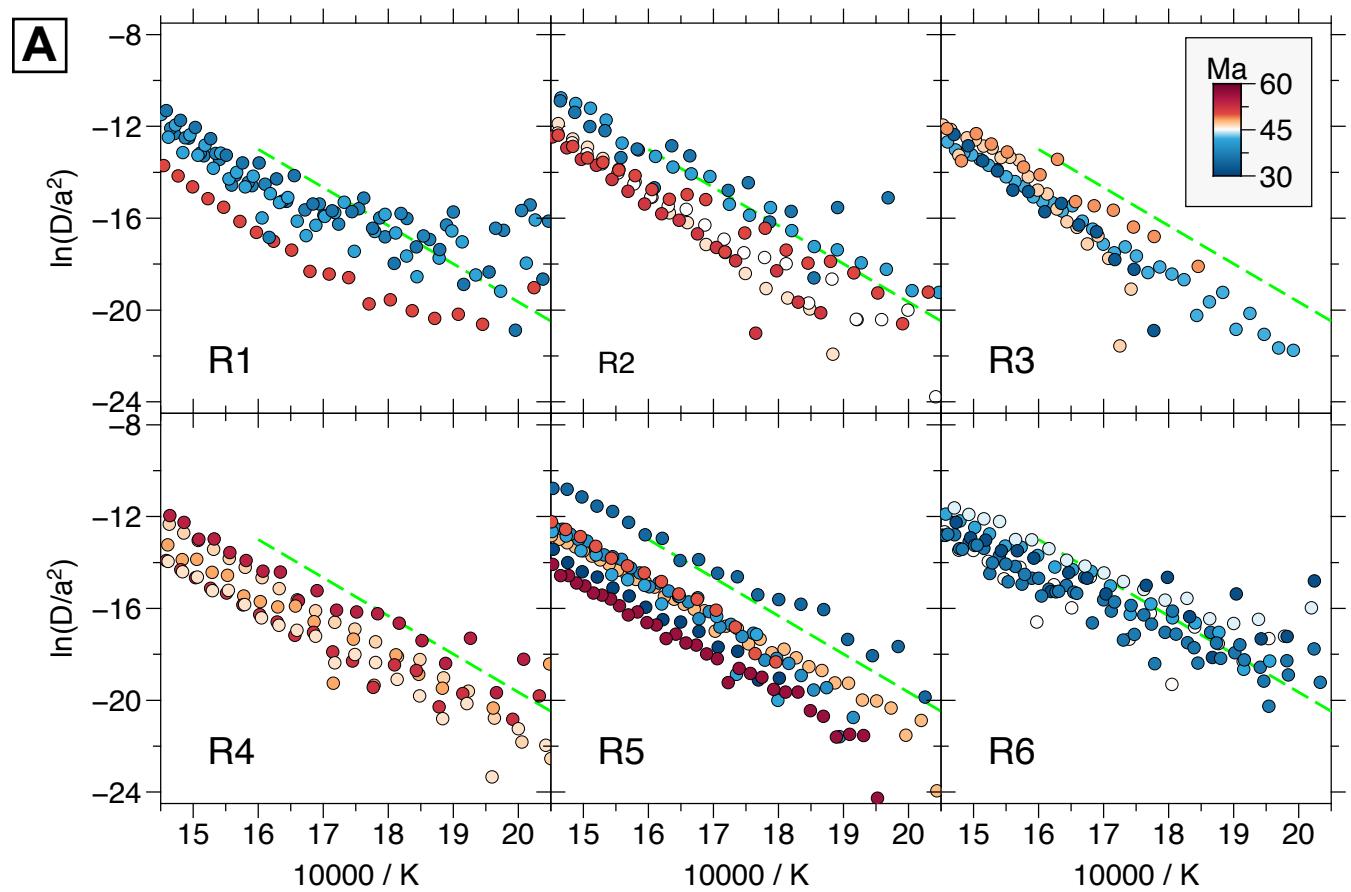


Figure 8

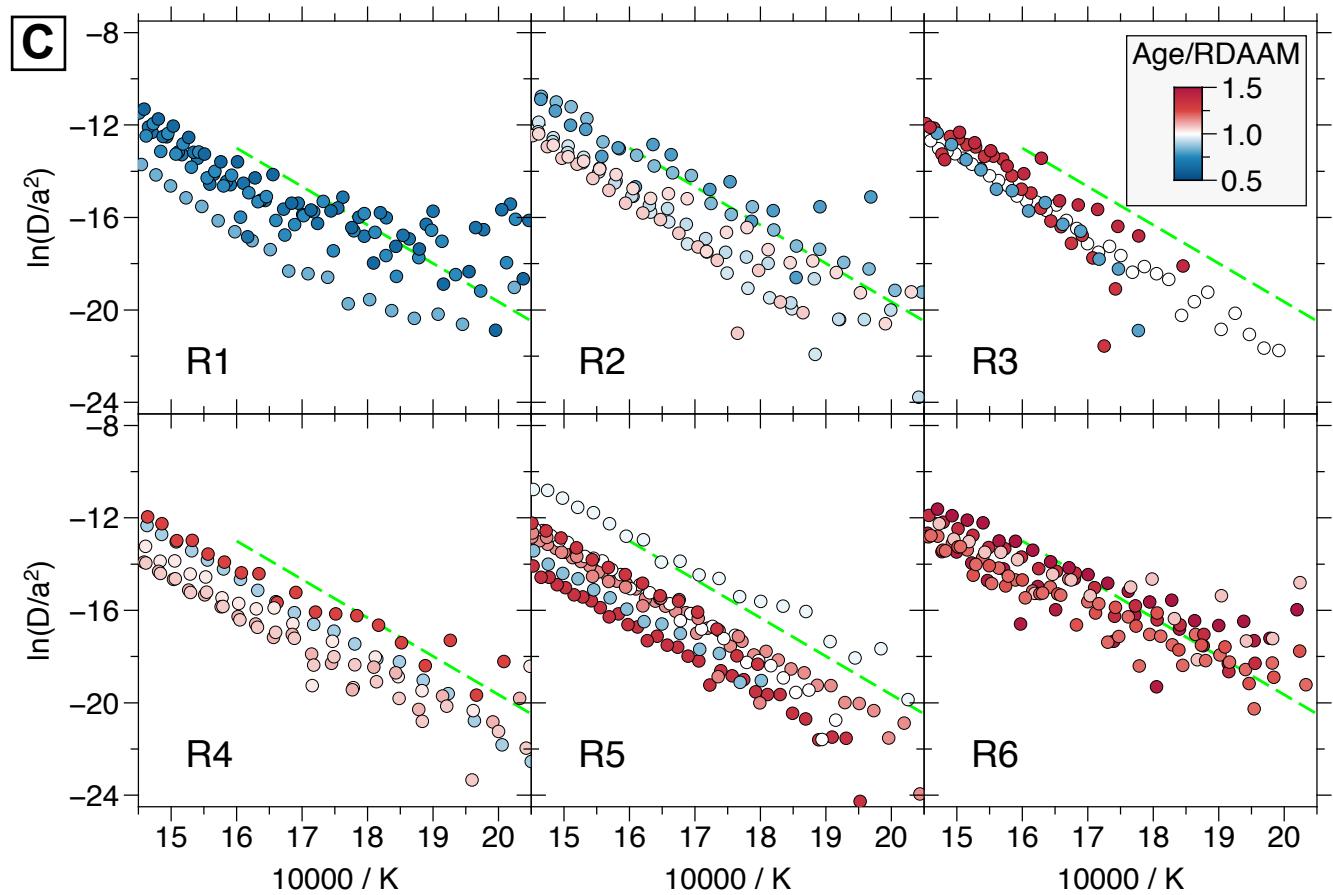


Figure 8 - continued

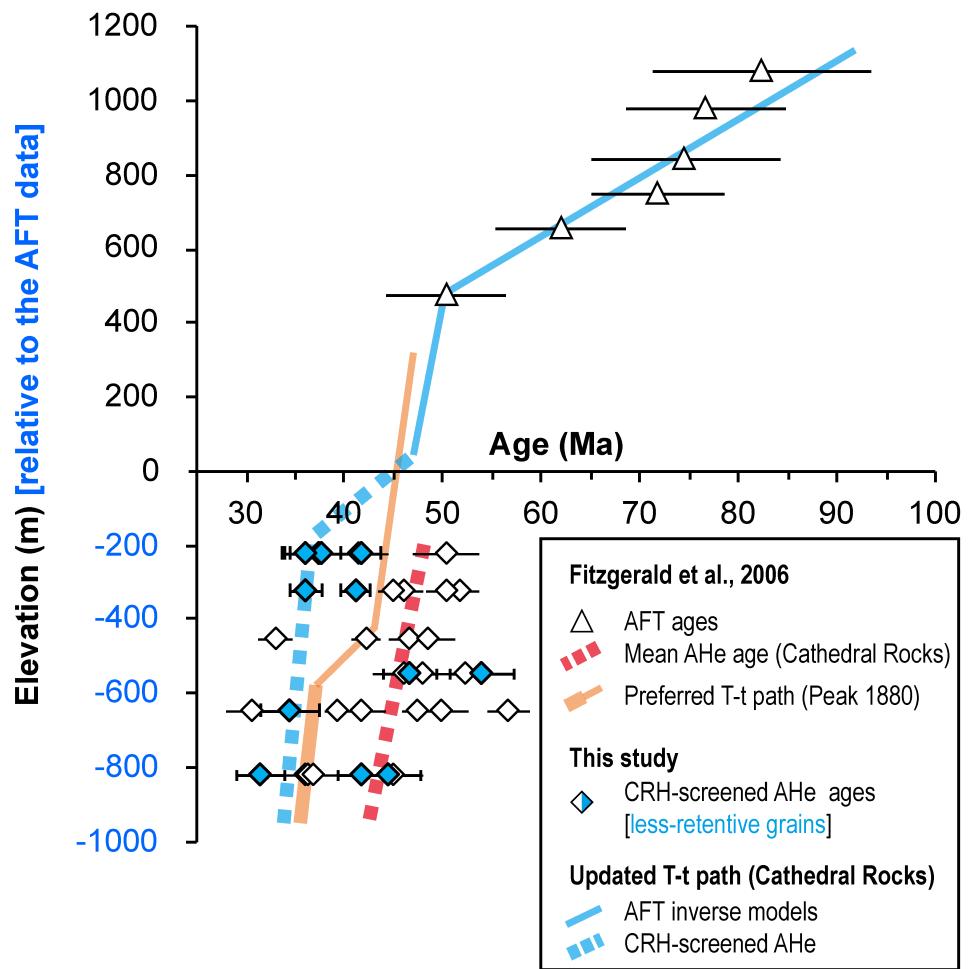


Figure 9

1 **Appendix A:** Supplementary figures with captions.

2 **Figure A1.** Ternary plot showing that compositions of TAM apatites are close to  
3 endmember fluorapatite, with little variation and minimal chlorine content. Points are  
4 color-coded according to rmr0 value (fission-track annealing parameter).

5 **Figure A2.** *df* curves for all samples. Large panels represent CRH analyses of each  
6 samples, in which the sub-plots represent single-grain CRH analyses. In each sub-plot,  
7 the single-grain sample name is marked at the upper left corner, and the single-grain  
8 total-gas age, CRH-corrected age, RDAAM-normalization, and RDAAM-normalized  
9 CRH-corrected age are listed at the upper right corner, respectively.

10 **Figure A3.** Arrhenius plots for all samples. Large panels represent Arrhenius plots of  
11 each samples, in which the sub-plots represent single-grain Arrhenius plots. In each sub-  
12 plot, the single-grain sample name, total-gas age, effective uranium, and Ft-equivalent  
13 spherical radius are listed at the upper right corner, respectively.

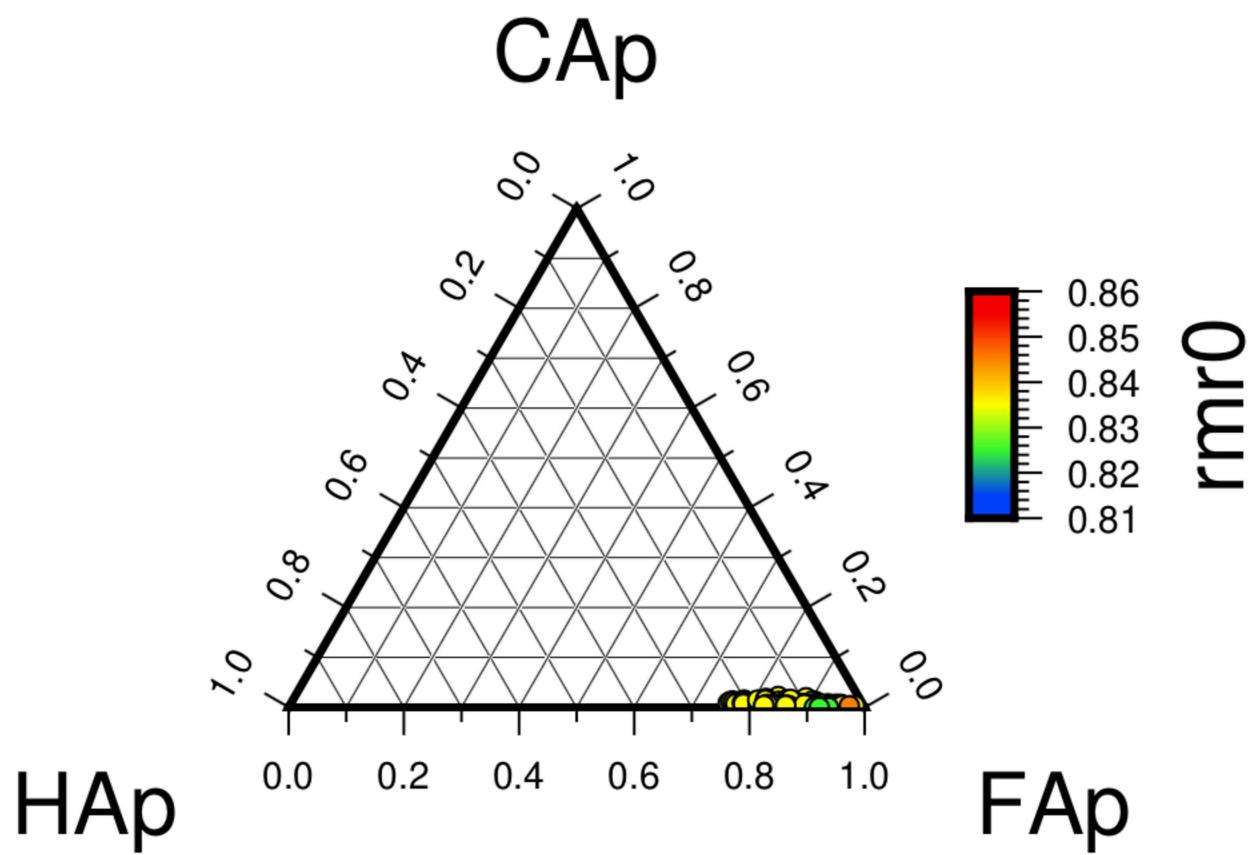


Figure A.1

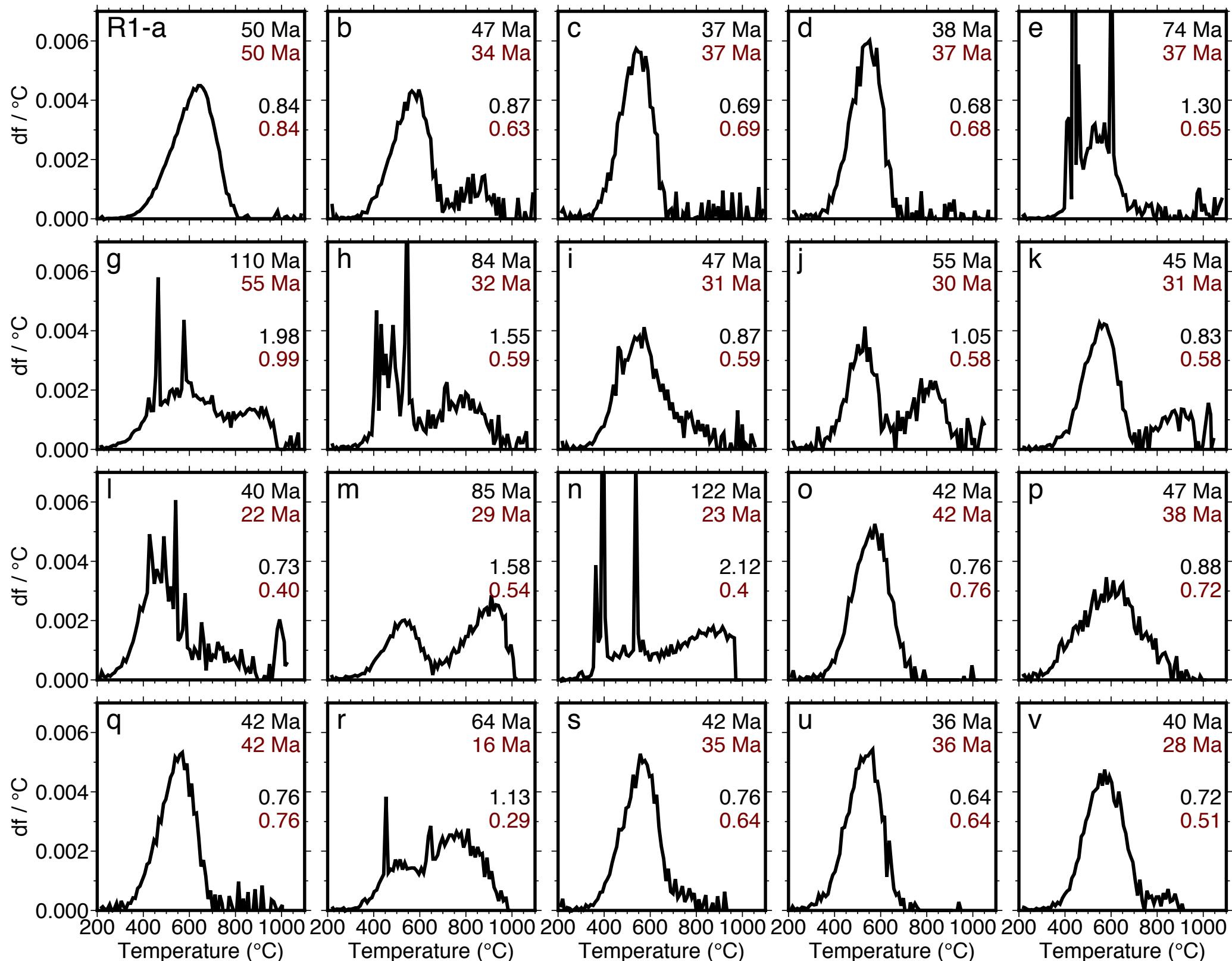
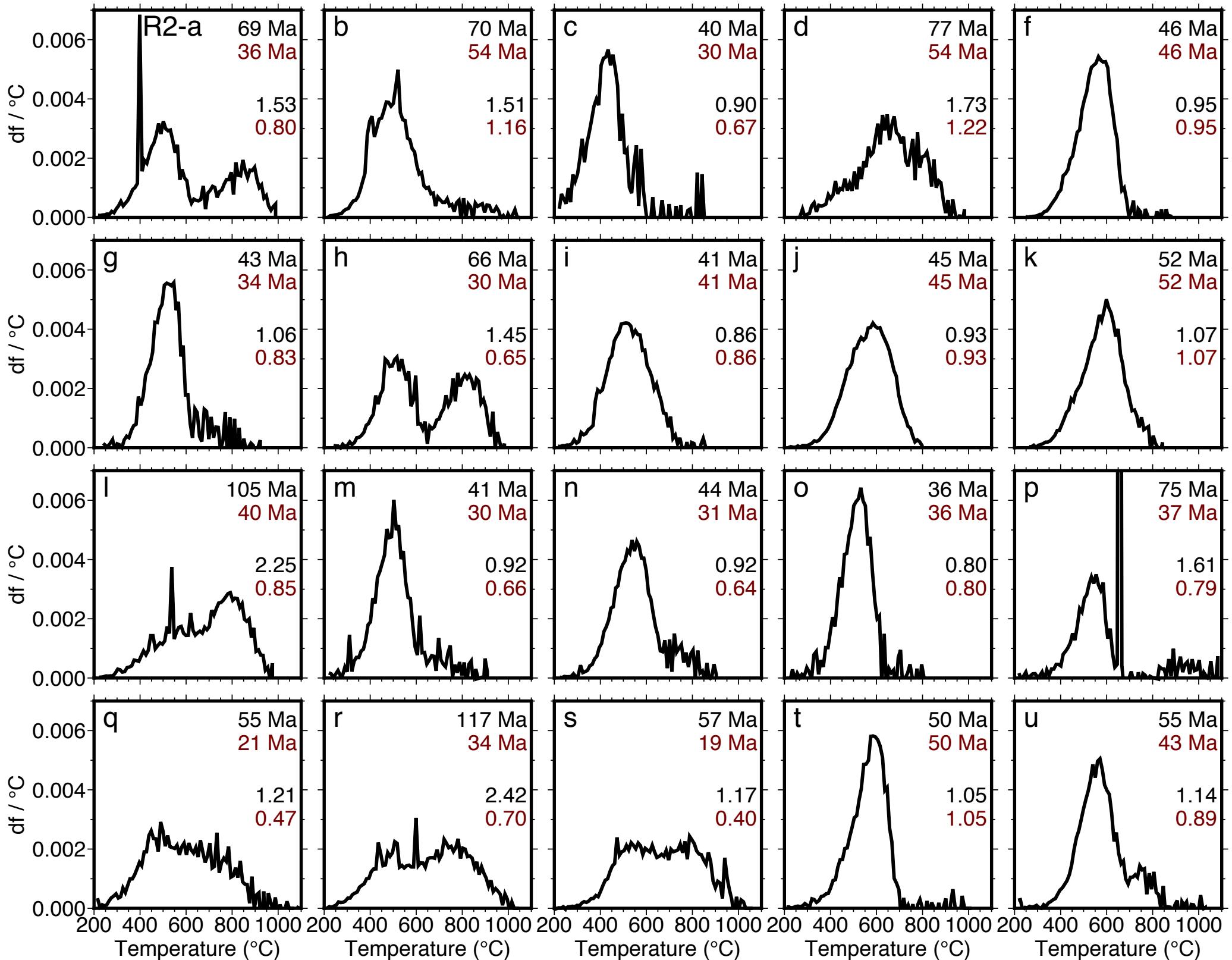


Figure A.2



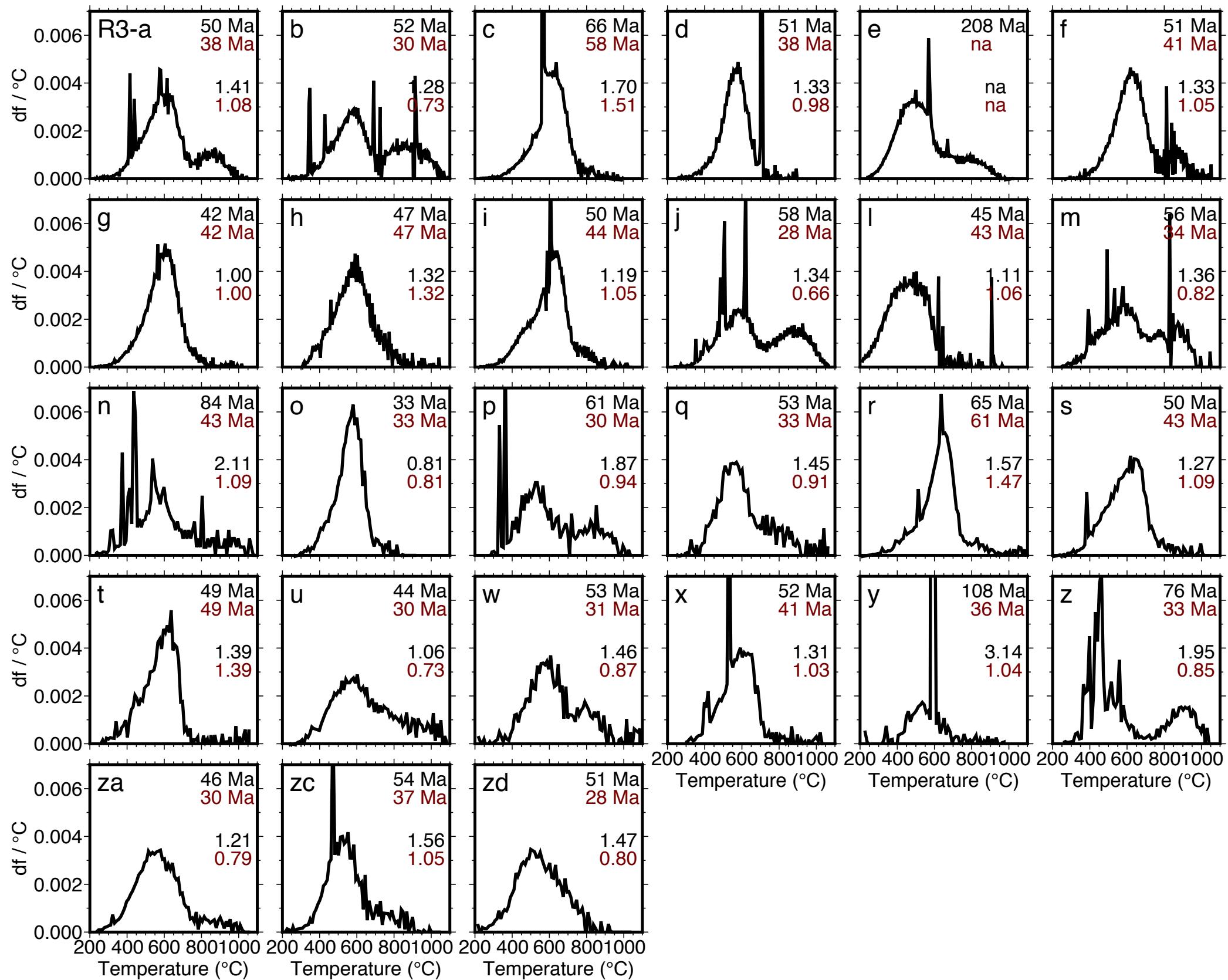


Figure A.2 - continued

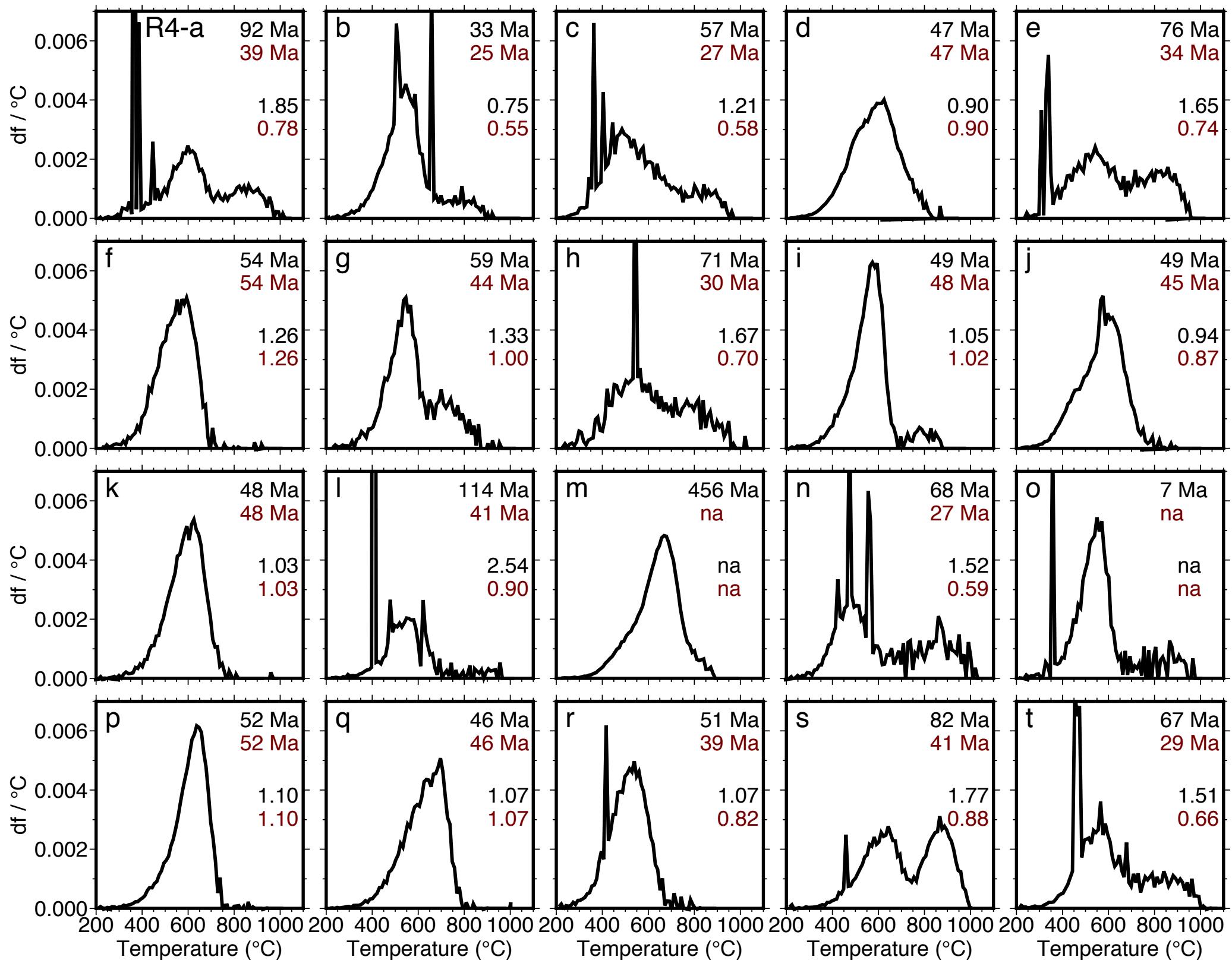


Figure A.2 - continued

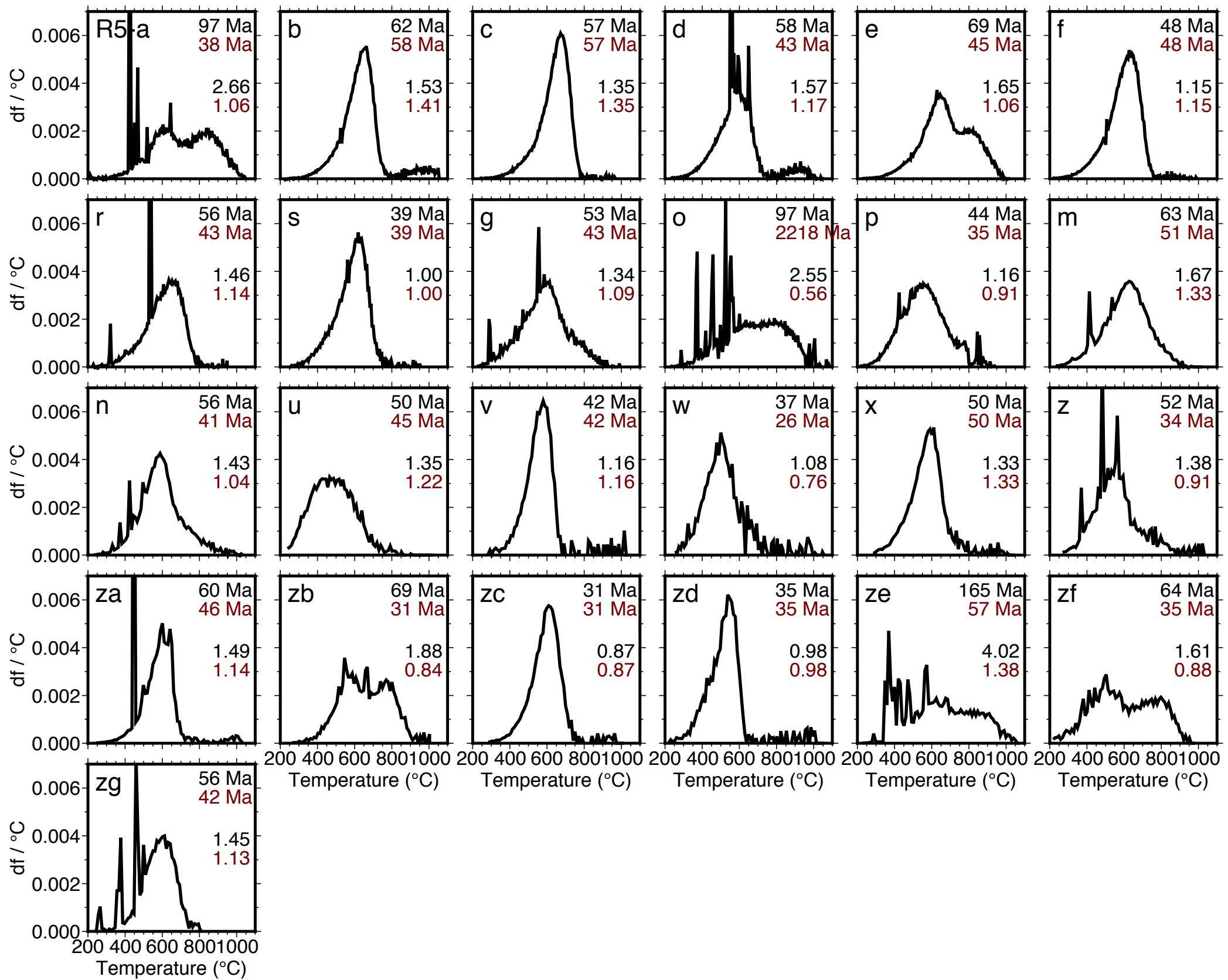
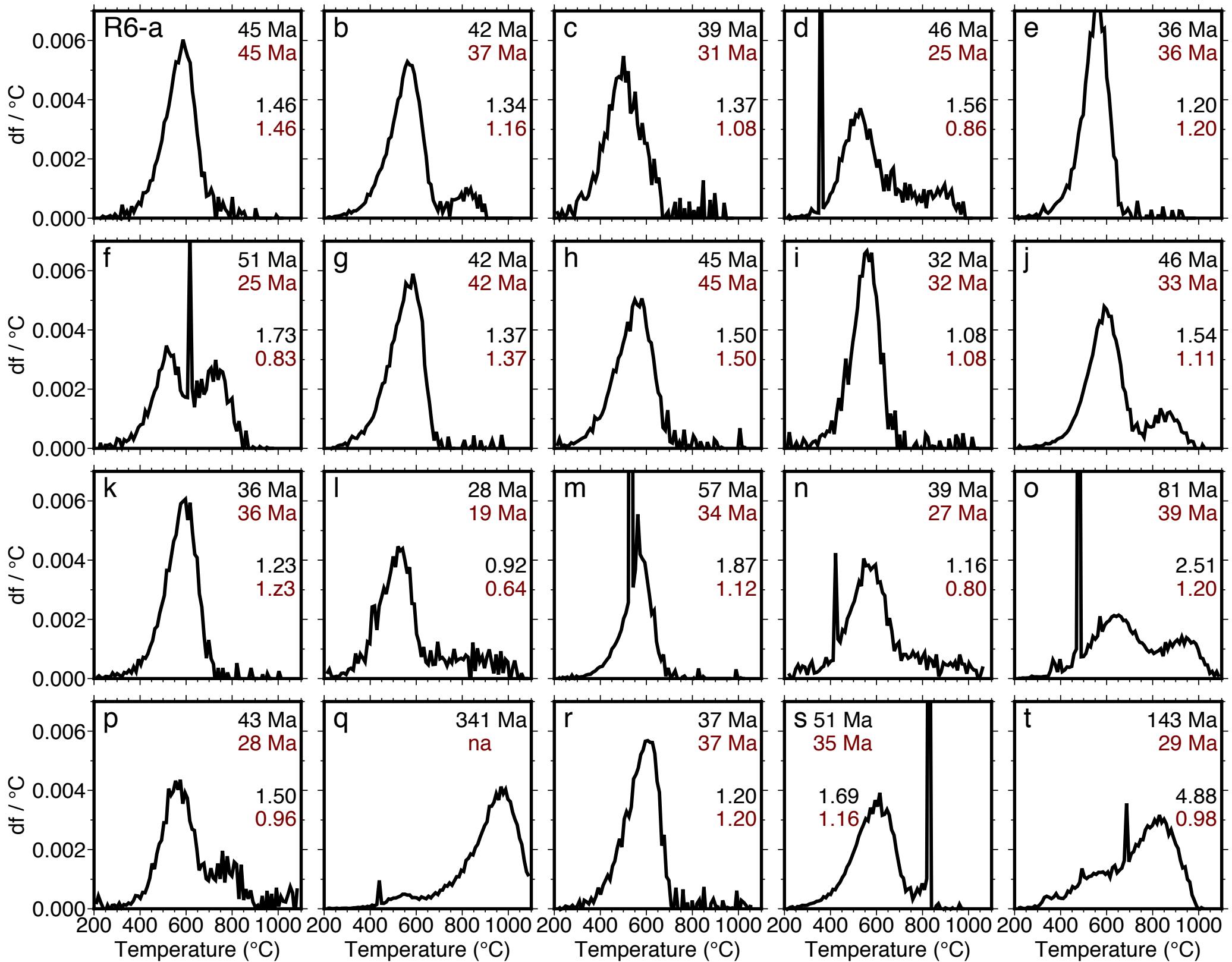


Figure A.2 - continued



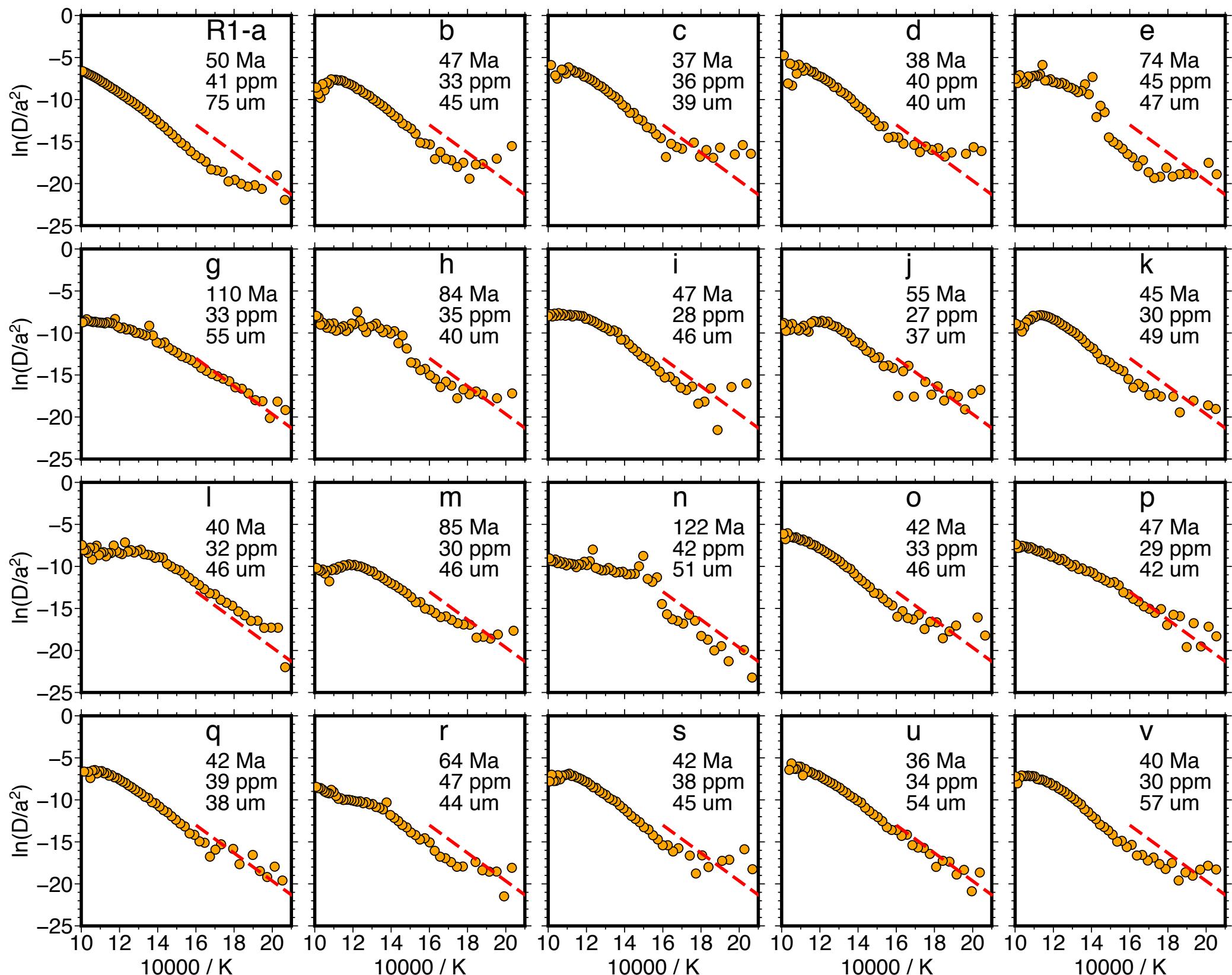
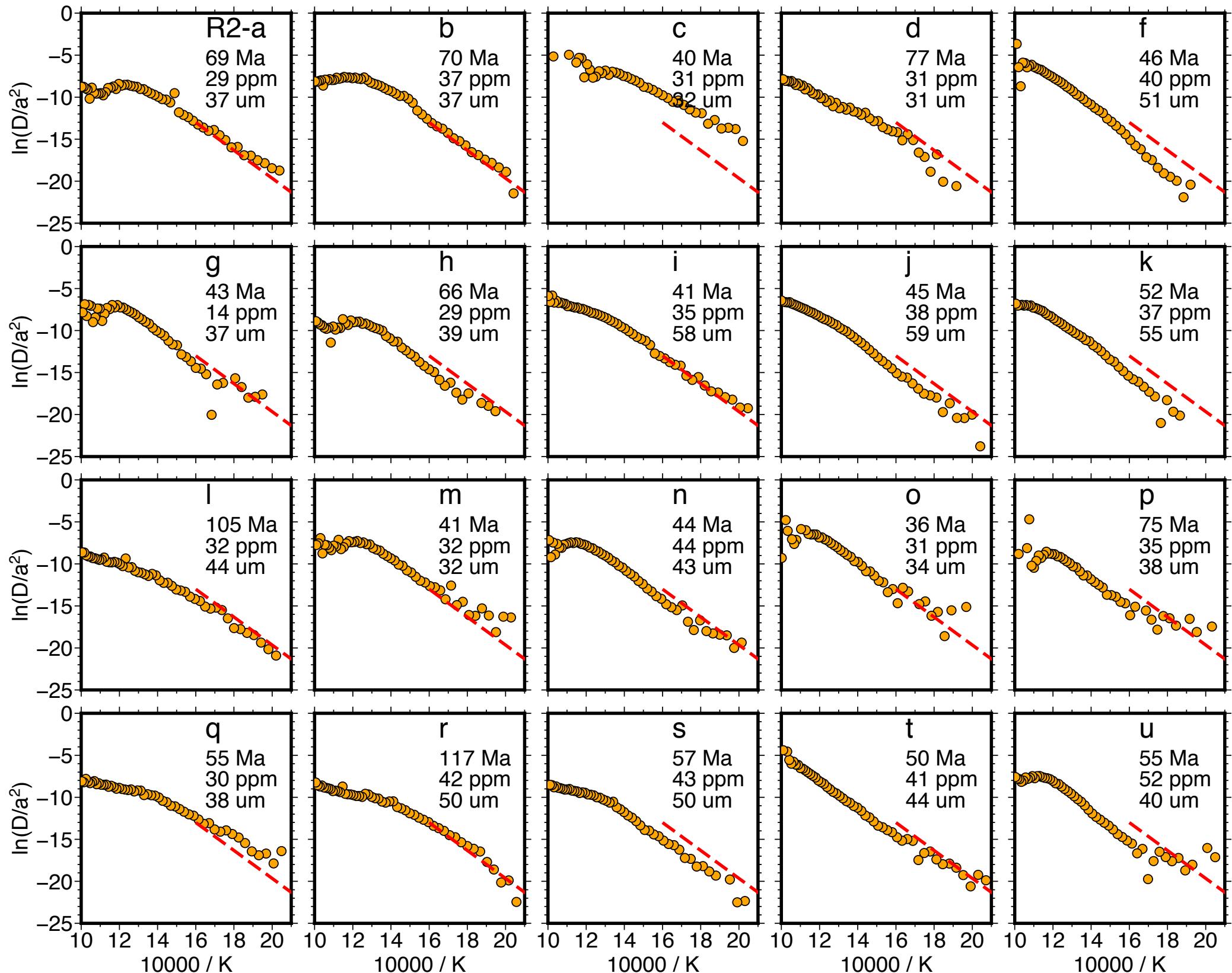
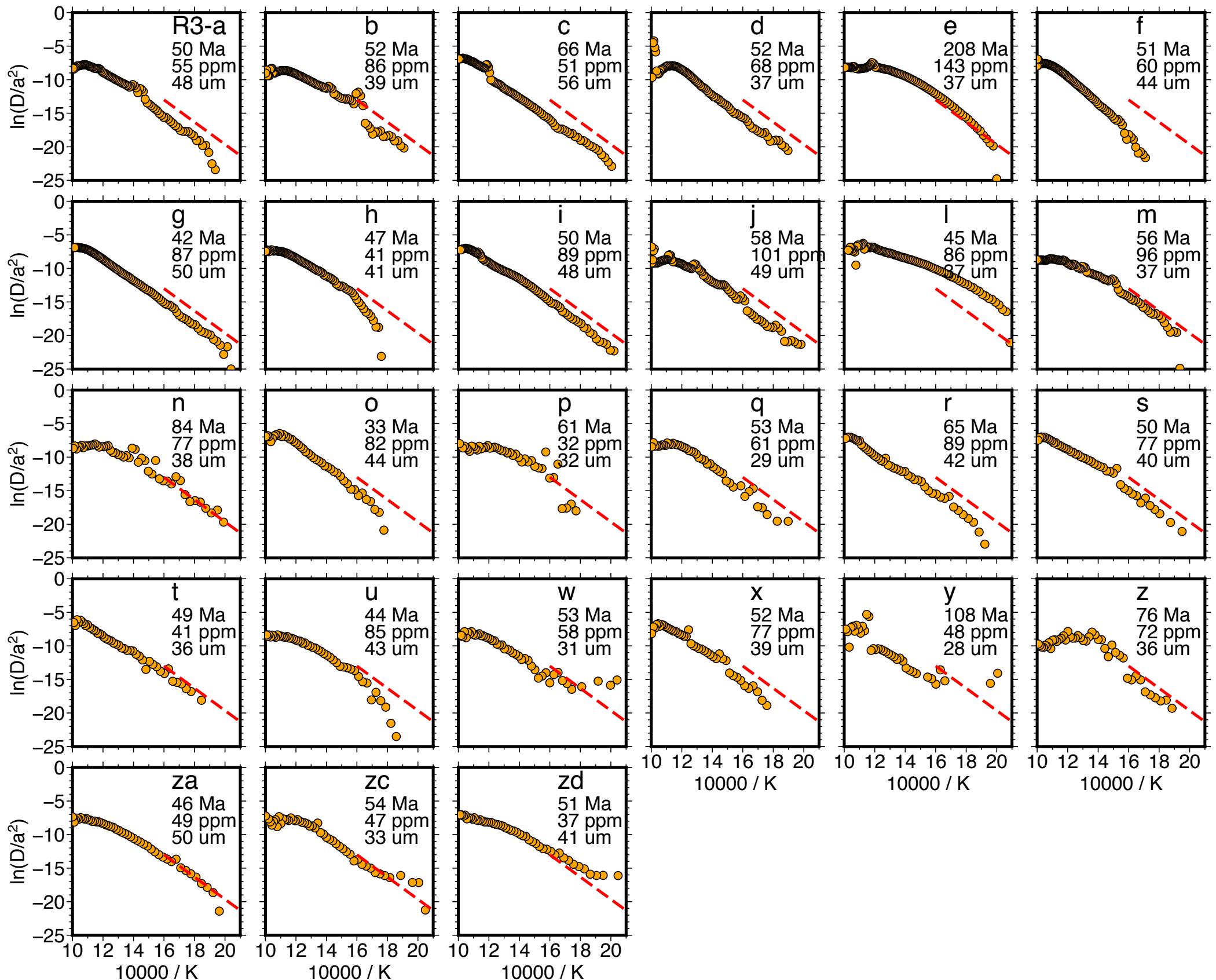
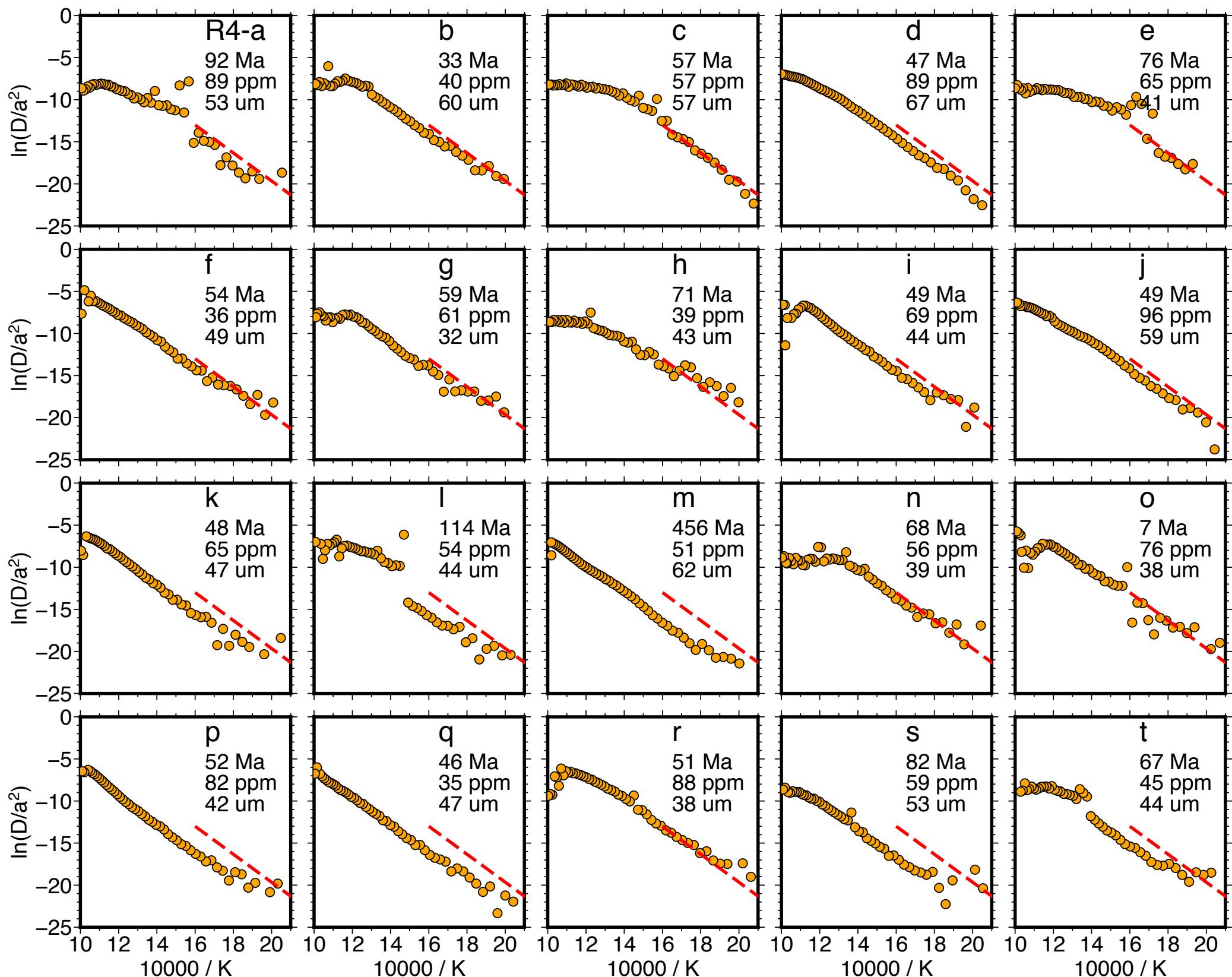


Figure A.3







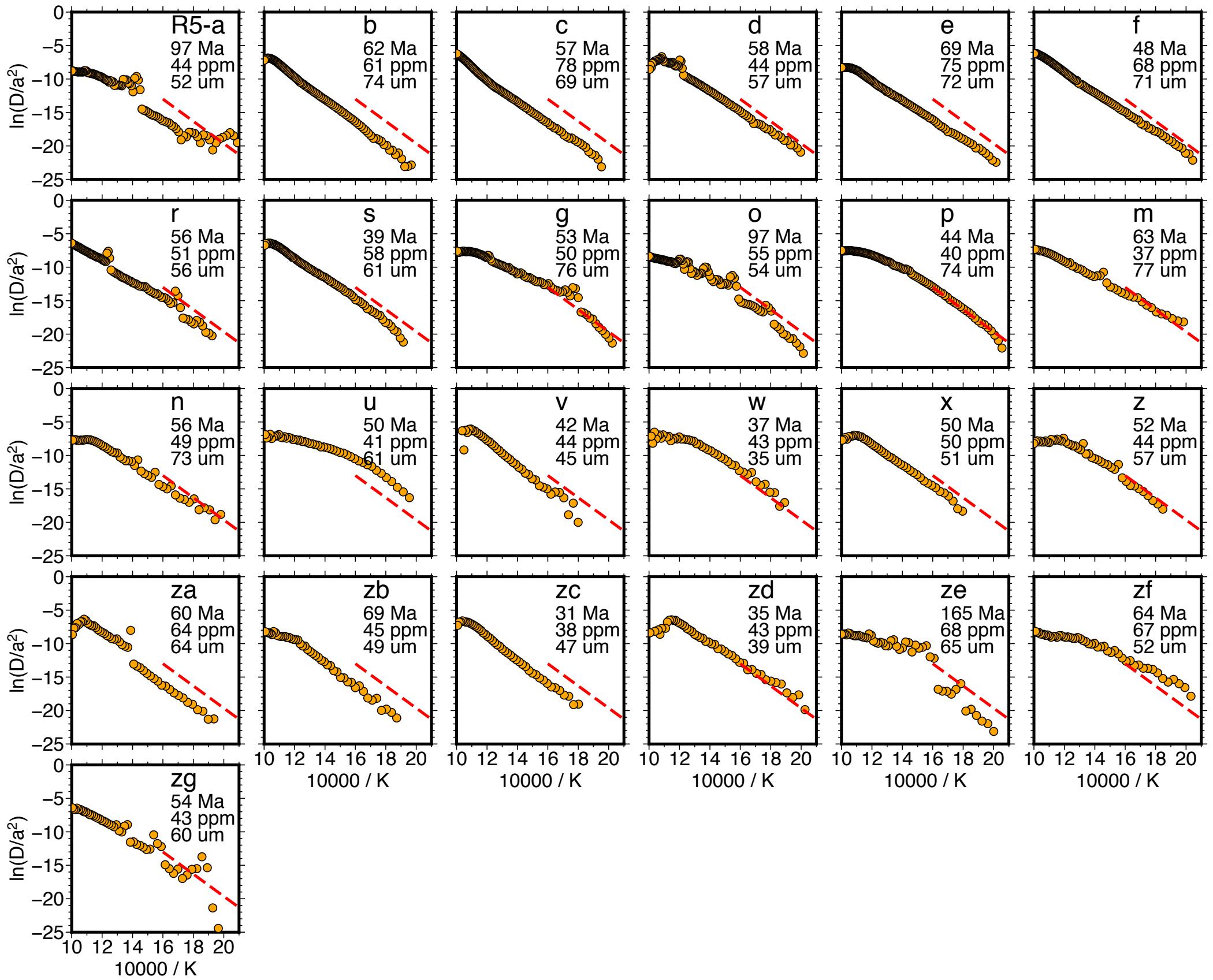
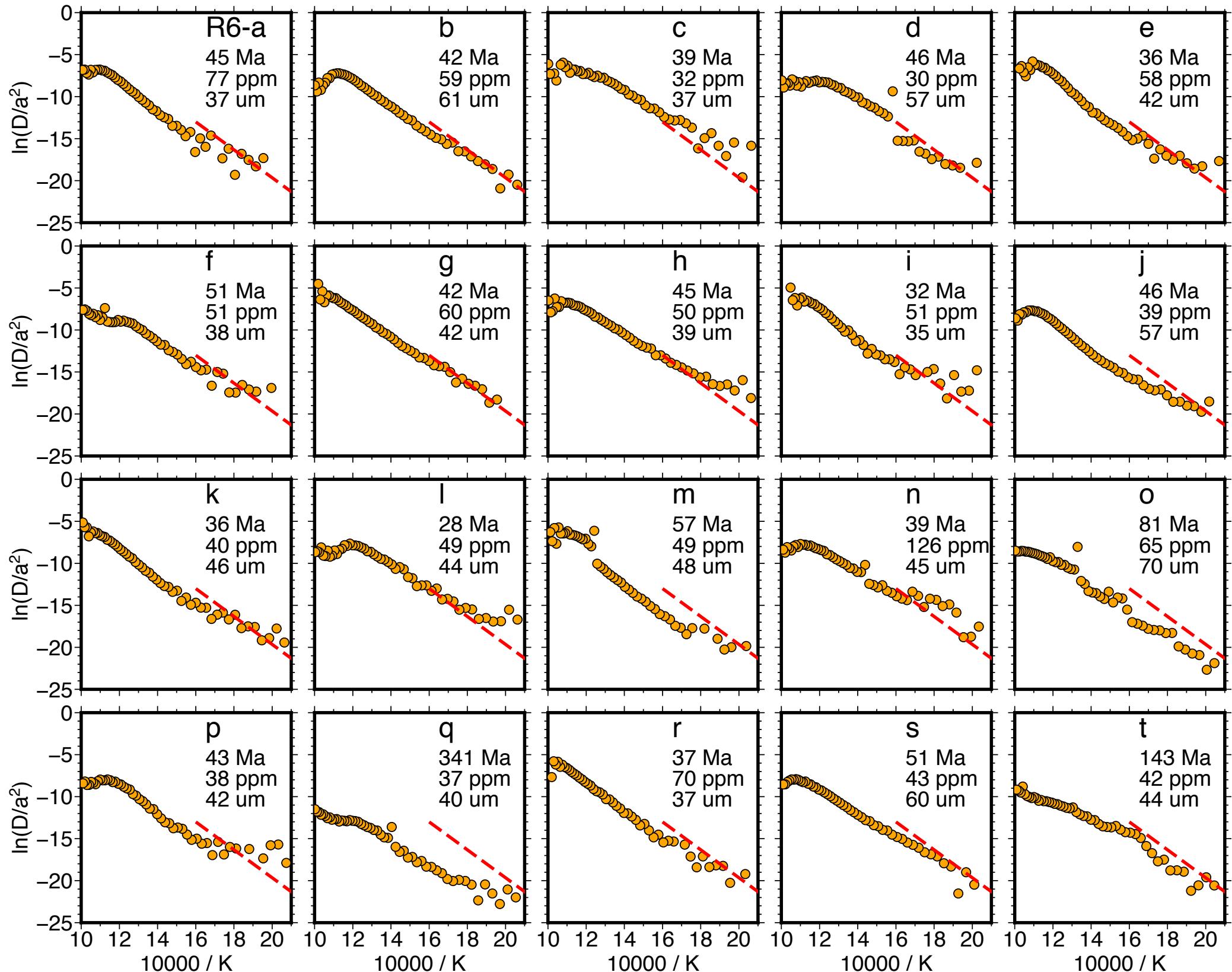


Figure A.3 - continued



1 **Appendix B: Documentation of key procedures for continuous ramped heating**

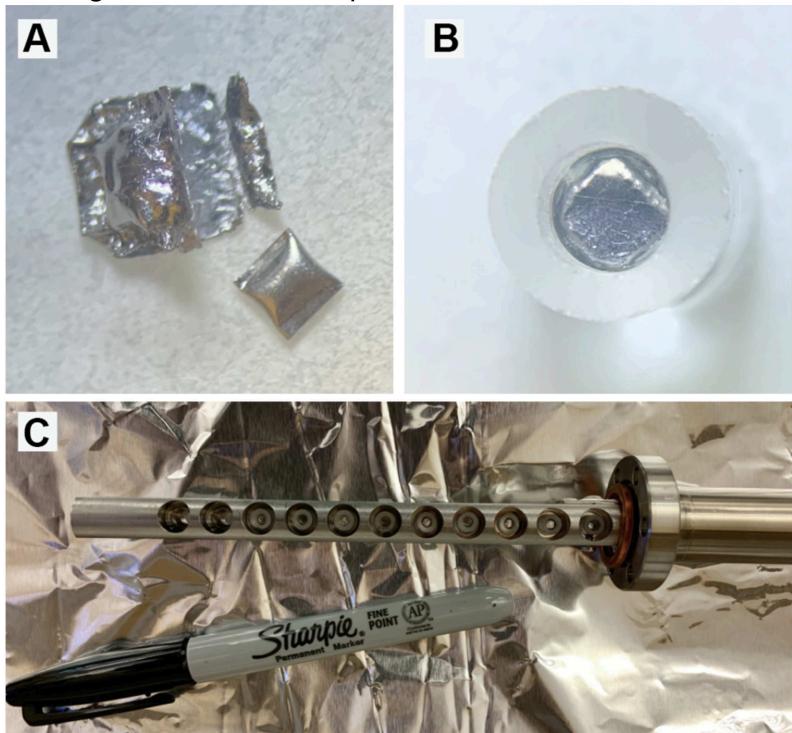
2

3 Details of the main analytical approach to Continuous Ramped Heating (CRH) can be  
4 found in Idleman et al. (2018). In this document we explain important updates to our  
5 analytical procedures and some caveats relevant to sample packaging and handling,  
6 laser heating and temperature measurement, gas gettering and monitoring, and data  
7 collection and reduction.

8

9 **1. Sample packaging and handling**

10 Each apatite grain, after being picked and photographed, was packaged in a niobium  
11 (Nb) tube with both ends closed – a standard approach for single-grain apatite (U-  
12 Th)/He dating. The tube was then wrapped in a Nb foil envelope (Fig. B1A). The Nb  
13 tubes we used for these experiments had an outside diameter of 0.4 mm and length of  
14 0.7 mm. The foil envelopes were octagonal-to-circle shaped with a diameter of ~2 mm,  
15 a size chosen to fit into the  
16 indentation in a cylindrical  
17 quartz glass sample holder  
18 (Fig. B1B) and also to  
19 completely contain the  
20 measurement spot of the  
21 optical pyrometer (~1.1 mm).  
22 The holders fit into a sample  
23 rack fitted with a linear



*Figure. B1: Sample packaging.*

24 actuator that allows up to 12 samples to be positioned under the laser beam (Fig. B1C).  
25 Both the Nb tubes and foil envelopes were cleaned with 7% HNO<sub>3</sub> and degassed for 3  
26 hours in a vacuum furnace at 600 °C before packaging. The overall goal of our  
27 packaging strategy is to keep the sample packet as small as possible in order to allow  
28 efficient and responsive laser heating (see following section) while still providing enough  
29 packet surface area for accurate temperature measurement.

30

## 31 **2. Laser heating and temperature measurement**

32 The TAM samples were heated with a 30 watt 808 nm fiber-coupled diode laser, and  
33 their temperatures were monitored using a BASF Exactus optical pyrometer positioned  
34 coaxially with the laser beam path as part of a custom beam delivery system. The  
35 pyrometer was recalibrated every 8-12 sample runs against a K-type thermocouple  
36 embedded in a “dummy” sample packet in one of sample-rack positions. Temperature  
37 regulation was achieved by modulating the laser power under closed-loop PID control  
38 using a custom Labview program. For the TAM experiments we performed CRH  
39 analysis using a heating rate of 30 °C/min. This rate was chosen to balance the  
40 responsiveness of the temperature control system, the speed of <sup>4</sup>He flow and pressure  
41 equilibration within the extraction line, the efficiency of active gas cleanup, and the time  
42 needed for the <sup>4</sup>He signal to be integrated and recorded with sufficient precision.

43

## 44 **3. Gas gettering and monitoring**

45 CRH analysis involves continuous sample heating and measurement of released helium  
46 gas under static conditions with the sample chamber open to the mass spectrometer.

47 The presence of other active gas components released from both the sample and the  
48 extraction system will impact CRH analysis and therefore requires careful attention to  
49 gettering and monitoring of these active gas contaminants. We have found that  
50 hydrogen and nitrogen in particular can alter the effective  ${}^4\text{He}$  sensitivity via pressure  
51 scattering or possibly other source effects. For the TAM experiments we used two  
52 SAES GP-50 getters operated at 20 and 200 °C in the extraction line for active gas  
53 cleanup, as described by Idleman et al. (2018). Fortunately, laser heating is particularly  
54 advantageous for CRH analysis, as the much smaller heated area relative to that  
55 required for furnace-heating experiments helps to minimize the evolution of these active  
56 gases. With our current laser heating setup, hydrogen and nitrogen co-evolved with  ${}^4\text{He}$   
57 have the greatest impact on the effective  ${}^4\text{He}$  sensitivity at temperatures where the  
58 sample is almost completely degassed (i.e., > 850 °C) and can result in slight (~1-5%)  
59 underestimations of the final  ${}^4\text{He}$  abundance. This problem can be addressed effectively  
60 by delaying the final measurement of the total  ${}^4\text{He}$  beam intensity for ~3 minutes after  
61 the termination of laser heating, allowing the partial pressures of the active gas  
62 contaminants to fall to levels at which their suppression of the  ${}^4\text{He}$  signal is trivial.

63

#### 64 **4. Data collection and reduction**

65 The  ${}^4\text{He}$  measurements were performed using a Pfeiffer Prisma Plus quadrupole mass  
66 spectrometer fitted with a channel electron multiplier. The CRH measurements were  
67 made in the static mode with the quadrupole fully open to the sample chamber.

68

69 The following sections highlight three key steps in our strategy for data collection and  
70 reduction:

71

72 **Before heating.** We allowed at least two minutes of measurement of the dynamic  
73 background in order to quantify the residual  $^4\text{He}$  signal present before the extraction line  
74 was valved off from its pumps. Once the pump valve was closed, we performed another  
75 two to three minutes of measurement of what we found to be a linearly rising time-  
76 dependent static  $^4\text{He}$  blank, allowing us to calculate the blank contribution over the  
77 course of the CRH experiment by extrapolation of these early measurements. In  
78 practice, we have found that the magnitude of this time-dependent static blank far  
79 exceeds the blank component contributed by sample heating, which in most  
80 experiments is trivial.

81

82 **During heating.** Temperature recording was initiated shortly before the start of laser  
83 heating. For the remainder of the experiment sample temperature and the  $^4\text{He}$  beam  
84 intensity were recorded at fixed intervals of 10 s and 10.2 s, respectively. During the first  
85 10-20 seconds of the experiment the samples were heated rapidly to  $\sim 160\text{-}170\text{ }^\circ\text{C}$   
86 under manual control to bring them within the measurement range of the pyrometer.  
87 After the sample temperature had stabilized for a few seconds control was passed to  
88 the Labview software, which then proceeded with the programmed heating schedule  
89 under closed-loop control. Heating and  $^4\text{He}$  measurement continued until the sample  
90 was completely outgassed, as indicated by either a stable  $^4\text{He}$  beam intensity for at

91 least one minutes or, more often, a slightly decreasing signal owing to the buildup of  
92 active gases.

93

94 **After heating.** After a sample was completely outgassed, we stopped temperature  
95 recording and allow the system to cool for three minutes, during which we commonly  
96 observed a 1-5% increase in the  ${}^4\text{He}$  beam current until the beam current finally  
97 stabilized. This increase typically coincided with an orders-of-magnitude drop in the m/e  
98 2 and 28 beam intensities as the sources of coevolved  $\text{H}_2$  and  $\text{N}_2$  cooled and these  
99 gases were gettered. We used the final  ${}^4\text{He}$  beam intensity achieved after this cleanup  
100 step for the age calculation. Once the  ${}^4\text{He}$  beam had been measured we introduced a  
101 calibrated aliquot of  ${}^4\text{He}$  as a standard addition to establish the  ${}^4\text{He}$  sensitivity of the  
102 quadrupole.

103

104 After completion of a CRH experiment, temperature and  ${}^4\text{He}$  abundance data were  
105 synchronized by linear interpolation using the time stamps recorded with each  
106 measurement. The accumulated  ${}^4\text{He}$  blank (static blank) was calculated using the  
107 previously determined accumulation rate and the heating duration. We then subtracted  
108 the dynamic background and the time-dependent static blank from our measured  ${}^4\text{He}$   
109 beam currents before calculating cumulative  ${}^4\text{He}$  loss as a function of temperature (f),  
110 and differentiation of the cumulative loss curve to obtain fractional loss (df/dT).