

# Metal–Organic Framework Based Hydrogen-Bonding Nanotrap for Efficient Acetylene Storage and Separation

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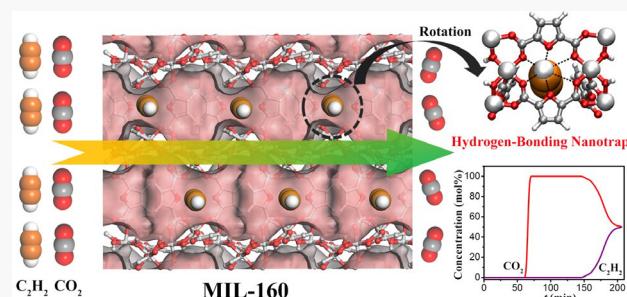
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**ABSTRACT:** The removal of carbon dioxide ( $\text{CO}_2$ ) from acetylene ( $\text{C}_2\text{H}_2$ ) is a critical industrial process for manufacturing high-purity  $\text{C}_2\text{H}_2$ . However, it remains challenging to address the tradeoff between adsorption capacity and selectivity, on account of their similar physical properties and molecular sizes. To overcome this difficulty, here we report a novel strategy involving the regulation of a hydrogen-bonding nanotrap on the pore surface to promote the separation of  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures in three isostructural metal–organic frameworks (MOFs, named MIL-160, CAU-10H, and CAU-23, respectively). Among them, MIL-160, which has abundant hydrogen-bonding acceptors as nanotraps, can selectively capture acetylene molecules and demonstrates an ultrahigh  $\text{C}_2\text{H}_2$  storage capacity ( $191 \text{ cm}^3 \text{ g}^{-1}$ , or  $213 \text{ cm}^3 \text{ cm}^{-3}$ ) but much less  $\text{CO}_2$  uptake ( $90 \text{ cm}^3 \text{ g}^{-1}$ ) under ambient conditions. The  $\text{C}_2\text{H}_2$  adsorption amount of MIL-160 is remarkably higher than those for the other two isostructural MOFs ( $86$  and  $119 \text{ cm}^3 \text{ g}^{-1}$  for CAU-10H and CAU-23, respectively) under the same conditions. More importantly, both simulation and experimental breakthrough results show that MIL-160 sets a new benchmark for equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  separation in terms of the separation potential ( $\Delta q_{\text{break}} = 5.02 \text{ mol/kg}$ ) and  $\text{C}_2\text{H}_2$  productivity ( $6.8 \text{ mol/kg}$ ). In addition, *in situ* FT-IR experiments and computational modeling further reveal that the unique host–guest multiple hydrogen-bonding interaction between the nanotrap and  $\text{C}_2\text{H}_2$  is the key factor for achieving the extraordinary acetylene storage capacity and superior  $\text{C}_2\text{H}_2/\text{CO}_2$  selectivity. This work provides a novel and powerful approach to address the tradeoff of this extremely challenging gas separation.



## INTRODUCTION

Gas separation and purification are crucial processes in the chemical industry to manufacture polymers, fuel, and plastics.<sup>1–4</sup> Acetylene ( $\text{C}_2\text{H}_2$ ) is not only an important gaseous fuel but also a fundamental building block for modern commodity chemicals.<sup>5</sup> However, impurity components (e.g.,  $\text{CO}_2$ ) are inevitably produced during the production of acetylene.<sup>6</sup> Since these two gas molecules have very similar molecular sizes/shapes ( $\text{C}_2\text{H}_2$ ,  $3.34 \times 3.32 \times 5.70 \text{ \AA}^3$ ;  $\text{CO}_2$ ,  $3.33 \times 3.18 \times 5.36 \text{ \AA}^3$ ) and physical properties (boiling points:  $\text{C}_2\text{H}_2$ ,  $189.3 \text{ K}$ ;  $\text{CO}_2$ ,  $194.7 \text{ K}$ ),<sup>7</sup> the separation of a  $\text{C}_2\text{H}_2/\text{CO}_2$  mixture is one of the most challenging separation tasks.<sup>8–10</sup> In recent years, a gas separation approach based on the physical adsorption of porous adsorbents has attracted extensive attention due to its low cost and energy-saving prospects,<sup>11–15</sup> in comparison to the traditionally energy intensive cryogenic distillation technology.<sup>16</sup>

The emerging metal–organic frameworks (MOFs)<sup>17–23</sup> feature high modularity, large surface areas, and abundant functionality, in comparison with traditional solid adsorbents (e.g., zeolite and activated carbon).<sup>24,25</sup> They can be readily constructed by combining metal ions/clusters with organic

linkers through coordination linkages<sup>26,27</sup> and have been well demonstrated to be promising in addressing many important gas separations (e.g., flue gas, olefin/paraffin).<sup>28–35</sup> It is a daunting challenge for a porous adsorbent in gas separation and purification to have both a high storage capacity and a high selectivity, the so-called tradeoff. For instance, MOFs with a high density of open metal sites (OMS; e.g., MOF-74 and HKUST-1) can improve the storage capacity of  $\text{C}_2\text{H}_2$ ; however, it has also been shown that a high affinity for  $\text{CO}_2$  can result in lower separation selectivity.<sup>36–39</sup> On the other hand, several ultramicroporous MOFs exhibit a high separation selectivity of  $\text{C}_2\text{H}_2/\text{CO}_2$  yet have a limited acetylene uptake capacity.<sup>40–43</sup> Therefore, there is an added value to develop an effective approach to address this tradeoff and achieve efficient separation of  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures.

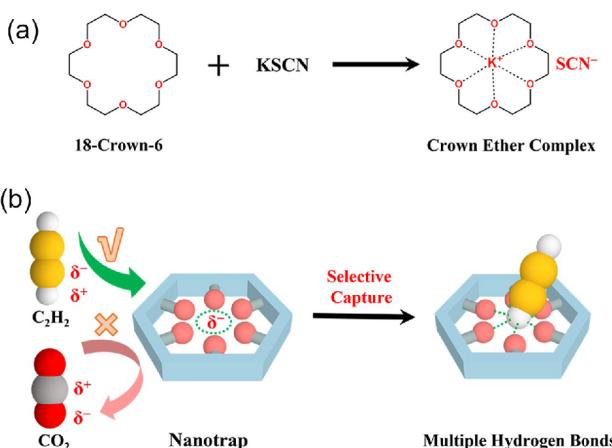
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Due to their nearly identical kinetic diameters (ca. 3.3 Å), effective separation of a  $\text{C}_2\text{H}_2/\text{CO}_2$  mixture cannot be achieved simply through fine-tuning the pore size, unlike the case for other gaseous mixtures (e.g.,  $\text{CO}_2/\text{CH}_4$ , alkyne/alkene).<sup>44,45</sup> Notably, there is a significant difference in the quadrupole moments and electrostatic potentials of carbon dioxide and acetylene ( $-13.4 \times 10^{-40}$  and  $+20.5 \times 10^{-40}$  C m<sup>2</sup> for  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$ ; Figure S1).<sup>46,47</sup> Thus, it is essential to fabricate specific functional sites within the porous MOFs that can preferentially bind with  $\text{C}_2\text{H}_2$  over  $\text{CO}_2$ . Inspired by the concept of specific recognition of crown ether complexes and macrocyclic molecules (Scheme 1a),<sup>48–50</sup> we hypothesize that

**Scheme 1. (a) Crown Ether Complex According to Pedersen<sup>48</sup> and (b) Selective Capture by a Hydrogen-Bonding Nanotrap of a Target Molecule through Host–Guest Multiple Hydrogen-Bonding Interactions**



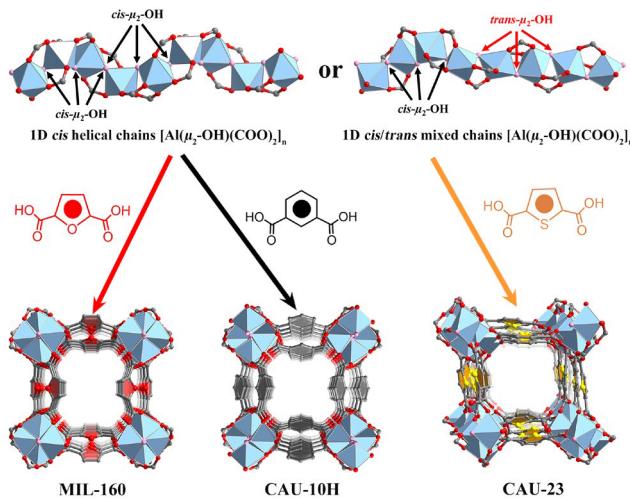
if the hydrogen-bonding nanotrap<sup>51</sup> (as acceptor) is immobilized on the pore surface of porous MOFs, it will not only provide strong hydrogen-bonding interactions to capture acetylene molecules (as donors) but also would not sacrifice the intrinsic pore volume (Scheme 1b).

In this regard, MIL-160, first reported by Serre and co-workers,<sup>52</sup> containing high-density hydrogen-bonding nanotraps within the pore surfaces, could be a promising candidate to achieve the aforementioned mission. To further demonstrate our strategy, here we synthesized three isostructural aluminum-based MOFs (named MIL-160, CAU-10H, and CAU-23, respectively)<sup>52–54</sup> on the basis of the isoreticular features of MOF chemistry and explored their binding abilities for  $\text{C}_2\text{H}_2$  and  $\text{C}_2\text{H}_2/\text{CO}_2$  separation selectivity for comparison. As expected, MIL-160 with the highest density hydrogen-bonding nanotraps has an excellent  $\text{C}_2\text{H}_2$  storage capacity ( $191 \text{ cm}^3 \text{ g}^{-1}$ , or  $213 \text{ cm}^3 \text{ cm}^{-3}$ ) but a much lower uptake of  $\text{CO}_2$  ( $90 \text{ cm}^3 \text{ g}^{-1}$ ) at 298 K and 100 kPa. Notably, the separation performance (storage capacity and selectivity) for  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures in MIL-160 is significantly higher than that of CAU-10H and CAU-23. Meanwhile, the simulation and experimental breakthroughs have fully demonstrated MIL-160 to be the best porous adsorbent reported thus far for equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  separation with regard to the separation potential ( $\Delta q_{\text{break}} = 5.02 \text{ mol/kg}$ ) and  $\text{C}_2\text{H}_2$  productivity (6.8 mol/kg). Moreover, computational modeling studies and *in situ* FT-IR tests both reveal that the ultrahigh acetylene storage capacity and superior  $\text{C}_2\text{H}_2/\text{CO}_2$  selectivity of MIL-160 can be mainly attributed to the unique host–guest multiple interactions

between the hydrogen-bonding nanotrap and  $\text{C}_2\text{H}_2$  molecule. This research provides a novel perspective to address the tradeoff for this challenging gas separation.

## RESULTS AND DISCUSSION

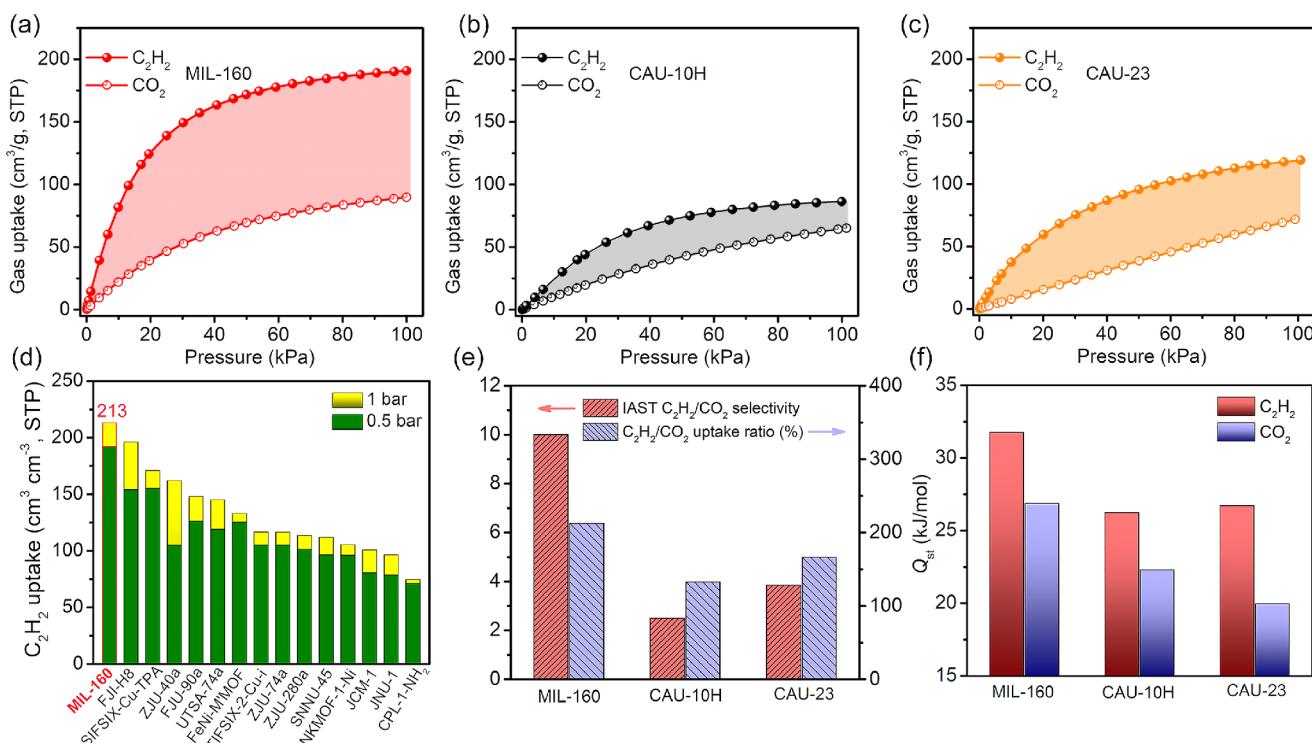
In these three isostructural MOFs MIL-160, CAU-10H, and CAU-23, each Al(III) atom is coordinated by six O atoms from two hydroxyl and four carboxylate groups, respectively, with a typical  $\text{AlO}_6$  octahedral geometry. Notably, in MIL-160 and CAU-10H, the two hydroxyl groups lie in *cis* positions, bridge the adjacent Al centers through a vertex-sharing mode, and form a 1D helical chain secondary building unit (SBU) along the *c* axis (Figure 1, top left). In contrast, another type of 1D



**Figure 1.** 1D chains  $[\text{Al}(\mu_2\text{-OH})(\text{COO})_2]_n$  and V-shaped ligands ( $\text{H}_2\text{FDC}$ ,  $\text{m-H}_2\text{BDC}$ , and  $\text{H}_2\text{TDC}$ ) assemble into three isostructural 3D frameworks of MIL-160, CAU-10H, and CAU-23, respectively. Hydrogen atoms are omitted for clarity. Color code, Al: pale blue; O, red and rose; C, 50% gray.

chain SBU in CAU-23 is formed by four consecutive alternating units of *trans* and *cis* corner-sharing  $\text{AlO}_6$  octahedra; this constructs a novel straight and helical mixed chain (Figure 1, top right). The above two types of 1D chains are quite different from the linear 1D chain with *trans*-connected mode in MIL-53 series of MOFs.<sup>55</sup> These 1D chains are further linked to four adjacent chains by the V-shaped ligands (FDC, m-BDC, and TDC) and extended into a three-dimensional (3D) open framework of MIL-160, CAU-10H, and CAU-23, respectively. After removal of the guest molecules, these three isostructural MOFs exhibit one-dimensional (1D) square channels, with pore aperture sizes of  $4.6 \times 9.8$ ,  $3.6 \times 9.2$ , and  $6.3 \times 7.6 \text{ \AA}^2$ , respectively (Figures S3–S5). Furthermore, some basic characteristics (e.g., FT-IR, TGA, and PXRD) of these three MOFs are provided in Figures S6–S14 (see the Supporting Information for details).

The permanent porous behaviors of these three MOFs were first investigated at 77 K with  $\text{N}_2$  adsorption experiments. All of them exhibit a fully reversible type I adsorption behavior, with Brunauer–Emmett–Teller (BET) surface areas of 1138, 680, and  $1320 \text{ m}^2 \text{ g}^{-1}$  for MIL-160, CAU-10H, and CAU-23, respectively (Figures S15–S17), which are consistent with the previously reported values.<sup>52–54</sup> In addition, the pore-size distributions (PSD) of these three MOFs were analyzed by employing the 77 K  $\text{N}_2$  isotherms according to the Horvath–



**Figure 2.** (a–c) Single-component  $\text{C}_2\text{H}_2$  and  $\text{CO}_2$  adsorption isotherms of MIL-160, CAU-10H, and CAU-23, respectively, at 298 K under 100 kPa. (d)  $\text{C}_2\text{H}_2$  volumetric uptake of MIL-160 in comparison to other best-performing MOF materials for  $\text{C}_2\text{H}_2/\text{CO}_2$  separation at room temperature. (e) Comparison of  $\text{C}_2\text{H}_2/\text{CO}_2$  selectivity and uptake ratio among these three MOFs at 298 K and 100 kPa. (f) Comparison of the heat of adsorption ( $Q_{st}$ ) of  $\text{C}_2\text{H}_2$  and  $\text{CO}_2$  at near-zero coverage for MIL-160, CAU-10H, and CAU-23.

Kawazoe cylinder model. All of them display narrow pore size distributions with main peaks at 7.0, 6.0, and 6.8 Å, respectively (Figure S18), consistent with the pore size determined by the crystal structure.

The permanent porosity and suitable pore size of these three MOFs provide us the initial motivation to explore their adsorption behaviors toward  $\text{C}_2\text{H}_2$  and  $\text{CO}_2$ . Therefore, the adsorption isotherms of  $\text{C}_2\text{H}_2$  and  $\text{CO}_2$  were collected at 273 and 298 K under 100 kPa pressure. As shown in Figure 2a–c, the  $\text{C}_2\text{H}_2$  uptake values at 298 K and 100 kPa are 191, 86, and 119  $\text{cm}^3 \text{g}^{-1}$  for MIL-160, CAU-10H, and CAU-23, respectively. Notably, the gravimetric  $\text{C}_2\text{H}_2$  uptake value ( $191 \text{ cm}^3 \text{g}^{-1}$ ) of MIL-160 is remarkably higher than those of the other two isostructural MOFs and is also superior to those of most of the top-performing materials for  $\text{C}_2\text{H}_2/\text{CO}_2$  separation, such as SIFSIX-Cu-TPA ( $185 \text{ cm}^3 \text{g}^{-1}$ ),<sup>56</sup> FJU-90a ( $180 \text{ cm}^3 \text{g}^{-1}$ ),<sup>57</sup> UTSA-74a ( $108 \text{ cm}^3 \text{g}^{-1}$ ),<sup>58</sup> FeNi-M' MOF ( $96 \text{ cm}^3 \text{g}^{-1}$ ),<sup>59</sup> Cu@FAU ( $79.5 \text{ cm}^3 \text{g}^{-1}$ ),<sup>60</sup> and Cu<sup>1</sup>@UiO-66-(COOH)<sub>2</sub> ( $52 \text{ cm}^3 \text{g}^{-1}$ )<sup>42</sup> and is only lower than that of the benchmark material FJI-H8 ( $224 \text{ cm}^3 \text{g}^{-1}$ ),<sup>61</sup> which has high-density open Cu sites under similar conditions (Table S6). In the actual separation process, the role of the volumetric storage capacity is even more important than the corresponding gravimetric capacity because it can make full use of the fixed-bed space and minimize the cost of energy regeneration. Under ambient conditions, MIL-160 exhibits the second highest volumetric  $\text{C}_2\text{H}_2$  adsorption capacity among the reported porous materials with a value of  $213 \text{ cm}^3 \text{cm}^{-3}$ . This value is only slightly lower than the record of  $230 \text{ cm}^3 \text{cm}^{-3}$  for CoMOF-74 with high-density OMS<sup>62</sup> and is significantly higher than those of other best-performing  $\text{C}_2\text{H}_2/\text{CO}_2$  separation materials (e.g.,  $196 \text{ cm}^3 \text{cm}^{-3}$  for FJI-H8,<sup>61</sup>  $147$

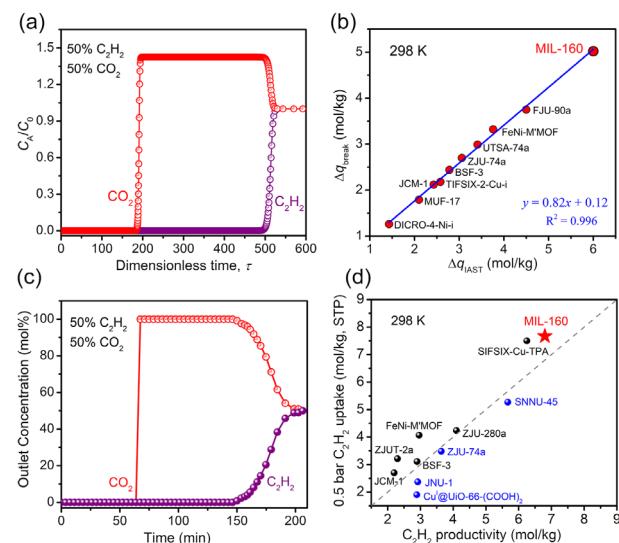
$\text{cm}^3 \text{cm}^{-3}$  for FJI-H8,<sup>57</sup> and  $145 \text{ cm}^3 \text{cm}^{-3}$  for UTSA-74a;<sup>58</sup> Figure 2d) under similar conditions. For equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  separation, the uptake value of  $\text{C}_2\text{H}_2$  at a partial pressure of 0.5 bar is another key indicator that needs to be considered. In this regard, the  $\text{C}_2\text{H}_2$  volumetric uptake of MIL-160 reaches up to  $192 \text{ cm}^3 \text{cm}^{-3}$  ( $7.7 \text{ mmol g}^{-1}$ ) at 298 K and 0.5 bar and is also superior to most of the best-performing materials for this separation (Figure 2d).

In addition, the density of the adsorbed acetylene in the channels can be determined as 518, 341, and 262 g/L for MIL-160, CAU-10H, and CAU-23 at 298 K and 100 kPa, respectively, on the basis of their  $\text{C}_2\text{H}_2$  adsorption capacities and corresponding pore volumes (Table S6). The packing density of  $\text{C}_2\text{H}_2$  in MIL-160 is notably higher than those of the other two isostructural MOFs and is also about 440 times the density of gaseous  $\text{C}_2\text{H}_2$  ( $1.1772 \text{ g/L}$ , at 273 K and 101.3 kPa) and is close to the density of solid  $\text{C}_2\text{H}_2$  at 189 K ( $729 \text{ g/L}$ ),<sup>63</sup> indicating that the acetylene molecules can be efficiently packed in the channels of MIL-160. On the other hand, MIL-160 shows the second highest safe acetylene storage density ( $0.247 \text{ g cm}^{-3}$ , slightly lower than the record of  $0.267 \text{ g cm}^{-3}$  for CoMOF-74; Table S6) in bulk material at 298 K and 100 kPa, which is about 100 times higher than the safe compression limit ( $0.2 \text{ MPa} = 0.0021 \text{ g cm}^{-3}$ ) of  $\text{C}_2\text{H}_2$  at room temperature.<sup>8</sup> For the practical application of acetylene storage, a promising adsorbent should have good repeatability and structural stability. Thus, we carried out five consecutive acetylene adsorption–desorption isotherm tests. As expected, there was no loss of  $\text{C}_2\text{H}_2$  storage capacity in MIL-160 after five cycles at 298 K, indicating that MIL-160 is a promising material in refillable  $\text{C}_2\text{H}_2$  storage (Figure S39).

In contrast, the  $\text{CO}_2$  uptake capacities of MIL-160, CAU-10H, and CAU-23 are only 90, 65, and  $72 \text{ cm}^3 \text{ g}^{-1}$ , giving  $\text{C}_2\text{H}_2/\text{CO}_2$  uptake ratios of 210%, 130%, and 160% (Figure 2e), respectively, at 100 kPa and 298 K. To further assess the separation performance of MIL-160, CAU-10H, and CAU-23 toward  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures, the separation selectivity was calculated by using the widely studied ideal adsorbed solution theory (IAST).<sup>64</sup> As shown in Figure S37b, the IAST selectivity of MIL-160 for equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures is up to 10 at 298 K and 100 kPa, which is about 2.6 and 4.0 times higher than the corresponding values in CAU-23 (3.8) and CAU-10H (2.5) and higher than those of some of the benchmark porous MOF materials, such as UTSA-74a (8.2),<sup>58</sup> FJU-90a (4.3),<sup>57</sup> MUF-17 (6.0),<sup>65</sup> SIFSix-21-Ni (7.8),<sup>66</sup> and TIFSix-2-Cu-i (6.5)<sup>67</sup> under similar conditions. Although some porous materials feature relatively higher  $\text{C}_2\text{H}_2/\text{CO}_2$  selectivity, their adsorption capacity of  $\text{C}_2\text{H}_2$  is relatively low (usually less than  $100 \text{ cm}^3 \text{ g}^{-1}$  under ambient conditions).<sup>42,43,68</sup> The high uptake ratio and IAST selectivity both suggested the potential of MIL-160 for effective  $\text{C}_2\text{H}_2/\text{CO}_2$  separation.

The affinity between the host framework and the guest molecule can be evaluated by the low-coverage heat of adsorption ( $Q_{st}$ ). The coverage-dependent  $Q_{st}$  values of MIL-160, CAU-10H, and CAU-23 for  $\text{C}_2\text{H}_2$  and  $\text{CO}_2$  were obtained by fitting the single-component gas isotherms collected at 273 and 298 K. As shown in Figure 2f, the near-zero coverage  $Q_{st}$  value of  $\text{C}_2\text{H}_2$  (31.8, 26.2, and 26.7  $\text{kJ mol}^{-1}$  for MIL-160, CAU-10H, and CAU-23, respectively) in these three isostructural MOFs is obviously higher than their corresponding values of  $\text{CO}_2$  (26.9, 22.3, and 20.0  $\text{kJ mol}^{-1}$ ), indicating that these MOFs have a stronger affinity for  $\text{C}_2\text{H}_2$  over  $\text{CO}_2$ . Unlike CAU-23, the  $Q_{st}$  value of  $\text{C}_2\text{H}_2$  in MIL-160 and CAU-10H gradually increases with an increase in  $\text{C}_2\text{H}_2$  loading (Figures S21, S24, and S27). This implies that  $\text{C}_2\text{H}_2$  molecules may form intermolecular interactions during the adsorption process,<sup>69–72</sup> which is consistent with the results of high-density  $\text{C}_2\text{H}_2$  storage in MIL-160 and CAU-10H. Additionally, the  $Q_{st}$  value of  $\text{C}_2\text{H}_2$  in MIL-160 is lower than those of some benchmark porous materials with high density of OMS, such as Cu-ATC (79.1  $\text{kJ mol}^{-1}$ ),<sup>10</sup> ZJU-74a (45  $\text{kJ mol}^{-1}$ ),<sup>73</sup>  $\text{Cu}^1@\text{UiO-66-(COOH)}_2$  (74.5  $\text{kJ mol}^{-1}$ ),<sup>42</sup> and  $\text{Cu}@\text{FAU}$  (50.0  $\text{kJ mol}^{-1}$ ).<sup>60</sup> The moderate  $Q_{st}$  value of  $\text{C}_2\text{H}_2$  indicates that MIL-160 can be regenerated under mild conditions, which give it a great potential for energy-efficient  $\text{C}_2\text{H}_2/\text{CO}_2$  separation.

To accurately evaluate the separation performance of porous materials in a fixed bed, we must consider not only the separation selectivity but also the adsorption capacity. To resolve this dilemma, a combined metric, termed the separation potential ( $\Delta q$ ), was defined by Krishna.<sup>74,75</sup> Subsequently, transient breakthrough simulations were performed for binary 50/50 or 90/10  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures in these three MOFs (Figure S40), operating at a total pressure of 100 kPa and 298 K, by employing the previously reported methodology.<sup>57,74–76</sup> As shown in Figure 3a, the equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures can be effectively separated by MIL-160, accompanied by the highest retention time ( $\Delta\tau = 304$ ; Table S2). Moreover, we compared the two types of separation potential ( $\Delta q_{break}$  and  $\Delta q_{IAST}$ ) of MIL-160 with those of other top-performing porous materials for  $\text{C}_2\text{H}_2/\text{CO}_2$  separation under ambient conditions (Figure 3b and Table S2).<sup>57–59,73,77</sup> It is worth noting that MIL-160 has not only the highest  $\Delta q_{IAST}$  (calculated on the basis of the static adsorption



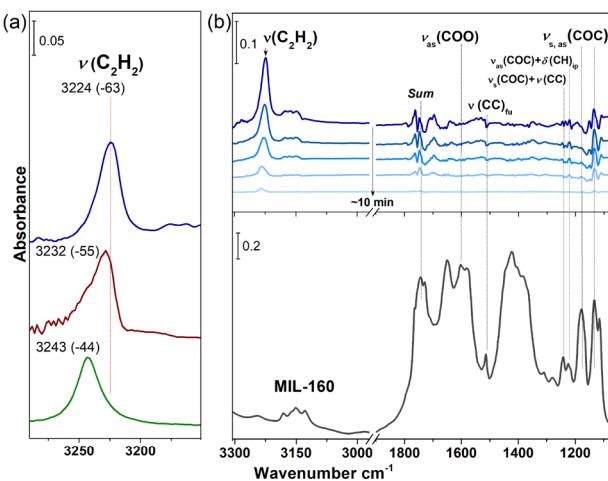
**Figure 3.** (a) Transient breakthrough curves of equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures for MIL-160 at 298 K and 100 kPa. (b) Comparison of the separation potential  $\Delta q$  of MIL-160 with those other benchmark porous materials. Note: the separation potentials  $\Delta q_{break}$  and  $\Delta q_{IAST}$  were calculated on the basis of the transient breakthrough simulations and the static adsorption isotherms for  $\text{C}_2\text{H}_2/\text{CO}_2$  (50/50) mixtures. (c) Experimental breakthrough curves for  $\text{C}_2\text{H}_2/\text{CO}_2$  (50/50) gas mixtures in a laboratory-scale fixed-bed packing with a MIL-160 sample under ambient conditions (298 K, 1 bar). (d) Single-component static  $\text{C}_2\text{H}_2$  uptake at a partial pressure of 0.5 bar and dynamic  $\text{C}_2\text{H}_2$  capture amount of equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures by MIL-160 in comparison with those of top-performing porous materials at room temperature (black balls, the dotted line, and blue balls indicate that 0.5 bar  $\text{C}_2\text{H}_2$  uptake is larger than, equal to, and less than dynamic  $\text{C}_2\text{H}_2$  capture, respectively).

isotherms) value of 6.0 mol/kg for equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures but also the highest  $\Delta q_{break}$  (5.02 mol/kg, calculated on the basis of the transient breakthrough simulations) value, indicating that MIL-160 has the best separation ability for this gas mixture. In addition, the separation potentials  $\Delta q_{break}$  and  $\Delta q_{IAST}$  have a good positive linear relationship ( $R^2 = 0.996$ ), with the equation  $y = 0.82x + 0.12$ . The linear range of this equation is approximately twice that of previously reported results.<sup>77,78</sup> Therefore, this empirical equation can be used to predict porous materials for challenging  $\text{C}_2\text{H}_2/\text{CO}_2$  separation after their  $\Delta q_{IAST}$  values have been calculated.

The record-high separation potential prompted us to evaluate the separation performance of MIL-160 for challenging  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures under real conditions. We performed laboratory-scale dynamic breakthrough experiments, with the equimolar  $\text{C}_2\text{H}_2/\text{CO}_2$  mixtures being flowed through a packed column filled with an activated sample ( $\sim 1.2$  g) at a total flow rate of  $2 \text{ mL min}^{-1}$  under ambient conditions. As shown in Figure 3c, the  $\text{C}_2\text{H}_2/\text{CO}_2$  mixture can be efficiently separated, in which the  $\text{CO}_2$  is eluted first at 64 min and rapidly reaches up to a pure grade without acetylene outflow, and this process continued for  $\sim 86$  min, a remarkable time to obtain pure  $\text{CO}_2$ , until the saturated uptake of acetylene and thus breakthrough (at 150 min). According to the breakthrough curve, the dynamic  $\text{C}_2\text{H}_2$  capture amount (also named productivity) was found to be 6.8 mol/kg for a given cycle, which is highly consistent with the equilibrium adsorption of  $\text{C}_2\text{H}_2$  under similar conditions (7.7 mol/kg, at 298 K and 0.5 bar). Notably, the dynamic  $\text{C}_2\text{H}_2$  productivity of

MIL-160 is among the highest values achieved and is much higher than those of most benchmark porous materials (Figure 3d, Table S3), such as FeNi-M' MOF (2.96 mol/kg),<sup>59</sup> ZJU-74a (3.64 mol/kg),<sup>73</sup> and Cu<sup>1</sup>@UiO-66-(COOH)<sub>2</sub> (2.89 mol/kg).<sup>42</sup> Therefore, MIL-160 establishes a new benchmark for the separation of challenging C<sub>2</sub>H<sub>2</sub>/CO<sub>2</sub> mixtures under actual conditions. Considering the need for recyclability in industrial applications, we carried out a multicycle mixed-gas breakthrough experiment under the same conditions. The results indicate no notable loss in the breakthrough time and C<sub>2</sub>H<sub>2</sub> capture capacity of MIL-160 in four consecutive cycles (Figures S41–S43), which proves that it maintains excellent repeatability for C<sub>2</sub>H<sub>2</sub>/CO<sub>2</sub> separation.

*In situ* infrared (IR) spectroscopic measurements were conducted to further probe the interaction of C<sub>2</sub>H<sub>2</sub> within three MOF structures: namely, MIL-160, CAU-23, and CAU-10H. C<sub>2</sub>H<sub>2</sub> is well-known to have an acidic nature and thus tends to form hydrogen bonds within the basic sites, as observed in several MOFs.<sup>34,79,80</sup> Similar to the well-studied OH stretching vibration,<sup>81</sup> the  $\nu_{as}(C_2H_2)$  band undergoes a downward shift with reference to the gas-phase value at 3287 cm<sup>-1</sup> when it is subjected to a hydrogen-bonding interaction and the frequency shift is a measure of intermolecular H bonds. Figure 4 presents the stretching band of adsorbed

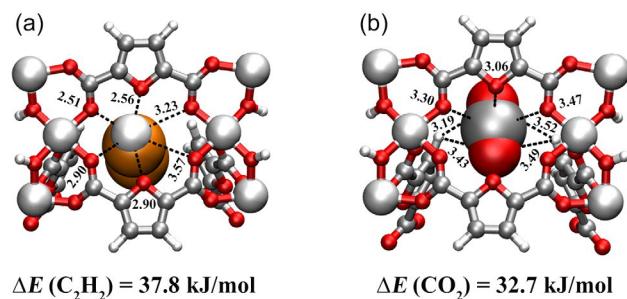


**Figure 4.** (a) IR spectra showing the stretching band of adsorbed C<sub>2</sub>H<sub>2</sub> inside MIL-160 (blue), CAU-23 (brown), and CAU-10 (green) upon loading C<sub>2</sub>H<sub>2</sub> gas at ~400 Torr (see full spectra in Figure 4b and Figure S45). The signal of gas-phase C<sub>2</sub>H<sub>2</sub> is subtracted. The values in parentheses show the shift of the  $\nu_{as}(C_2H_2)$  band with respect to the gas phase at 3287 cm<sup>-1</sup>. (b) The top five difference spectra showing the unloading of C<sub>2</sub>H<sub>2</sub> from MIL-160 upon evacuation of the gas phase for ~10 min; each is referenced to the bottom spectrum of activated MIL-160 under vacuum. Notation and acronyms:  $\nu$ , stretch;  $\delta$ , deformation; as, asymmetric; s, symmetric; ip, in plane; sum, summation band; fu, furan ring.

acetylene upon loading C<sub>2</sub>H<sub>2</sub> at ~400 Torr into three MOFs and its desorption from MIL-160 after evacuating the gas phase. As shown in Figure 4a,  $\nu_{as}(C_2H_2)$  bands occur at 3224, 3232, and 3243 cm<sup>-1</sup> in MIL-160, CAU-23, and CAU-10H, respectively. The trend of band position shift indicates that hydrogen-bonding interactions follow the order MIL-160 > CAU-23 > CAU-10H. In addition, different spectra shown in Figure 4b show that vibrational modes associated with the organic linker, including the COC symmetric and asymmetric stretch at 1178 and 1133 cm<sup>-1</sup>, the coupled COC stretch/CC

stretch/in-plane C–H deformation at 1242–1224 cm<sup>-1</sup>, the furan ring carbon stretch at 1514 cm<sup>-1</sup>, the carboxylate asymmetric stretch at 1580–1602 cm<sup>-1</sup>,<sup>82</sup> and the out-of-plane C–H summation band at 1743 cm<sup>-1</sup>,<sup>83</sup> are appreciably perturbed upon loading acetylene into MIL-160 (see assignment in Table S4). With desorption of the trapped acetylene from MIL-160, the perturbations of these modes gradually disappeared, as seen in Figure 4b. These observations point to the direct interaction of loaded C<sub>2</sub>H<sub>2</sub> molecules with both COO- and the O-containing ring of the FDC linker, as further verified by our modeling studies (Table S4).

To gain insight into the binding sites of C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> in MIL-160, modeling studies based on simulated annealing calculations and canonical Monte Carlo (CMC) simulations were performed (see the Supporting Information for details). As shown in Figure S46, three and two possible binding sites of C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> were found in MIL-160, respectively. Here, we focus on discussing and comparing the primary binding site of C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> in MIL-160, and discussions on other binding sites are provided in the Supporting Information. It was observed that the orientation of the C<sub>2</sub>H<sub>2</sub> (I) molecule is roughly perpendicular to the MOF channel (Figure S46a), where it forms multiple H–C≡C–H<sup>δ+</sup>...O<sup>δ-</sup> hydrogen-bonding interactions with the nearby carboxylate groups and furan rings, forming a nanotrap on the pore surface, with close interaction distances ranging from 2.51 to 3.57 Å (Figure 5a).



**Figure 5.** Canonical Monte Carlo (CMC) simulated primary binding sites of (a) C<sub>2</sub>H<sub>2</sub> and (b) CO<sub>2</sub> in MIL-160. Distances are given in Å.

In contrast, the CO<sub>2</sub> (I) molecule is oriented parallel to the *c* axis in the channels (Figure S46b),<sup>84</sup> in which (CO<sub>2</sub>)C<sup>δ+</sup>...O<sup>δ-</sup> electrostatic interactions with the carboxylate groups and the furan O atom (distances ranging from 3.06 to 3.47 Å) and CH<sup>δ+</sup>...O<sup>δ-</sup>(CO<sub>2</sub>) interactions with the nearby furan rings (distances ranging from 3.19 to 3.52 Å) were observed (Figure 5b). Overall, C<sub>2</sub>H<sub>2</sub> exhibits typically shorter interaction distances with MIL-160 at the primary binding site in comparison to CO<sub>2</sub>, which accounts for the stronger affinity of C<sub>2</sub>H<sub>2</sub> with the adsorbent. The calculated binding energies (37.8 vs 32.7 kJ mol<sup>-1</sup>) for single C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> molecules positioned at their global minimum in MIL-160 from CMC simulations were consistent with the trend in the experimental low-coverage  $Q_{st}$  values (31.8 vs 26.9 kJ mol<sup>-1</sup>) from single-component adsorption isotherms (Table S5). In summary, the binding configurations in MIL-160 calculated herein combine key features that cause stronger C<sub>2</sub>H<sub>2</sub> binding versus CO<sub>2</sub>: multiple H–C≡C–H<sup>δ+</sup>...O<sup>δ-</sup> hydrogen-bonding interactions and weak CO<sub>2</sub>–sorbent electrostatic interactions.

The diffusion of C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> molecules inside MIL-160 was investigated by looking at various transition-state configurations of the molecules traveling through the MOF

channel using the climbing nudged elastic band (cNEB) method.<sup>85,86</sup> To better understand the interaction taking place during molecular diffusion, induced charge densities were mapped for the initial, transition, and final states, as shown in Figure S48. For the initial configuration (at 0% progress), both C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> molecules occupy the space near their respective primary binding sites, giving the lowest energy to this configuration. Subsequently, the gas molecules move along the channel, getting to the highest energy configuration with diffusion barriers of 157 meV for C<sub>2</sub>H<sub>2</sub> and 161 meV for CO<sub>2</sub>, respectively. At the initial state, C<sub>2</sub>H<sub>2</sub> and CO<sub>2</sub> show interactions with the MOF linkers, as evidenced by the yellow/blue induced charge densities in Figure S48. However, at the transition configuration CO<sub>2</sub> shows a reduced interaction with the surrounding linkers, as evidenced by the reduction in induced charge density. In contrast, C<sub>2</sub>H<sub>2</sub> still exhibits strong interaction with the linker atoms (see Figure S48a for the transition state). This results in a transition configuration that is more stable for C<sub>2</sub>H<sub>2</sub> in comparison to CO<sub>2</sub>. Therefore, the diffusion energy barrier of C<sub>2</sub>H<sub>2</sub> is slightly lower, leading to a faster molecular transport along the MOF channel, indicating that less energy is required for acetylene molecules to leave the strong binding site and diffuse along the MOF channel. The higher binding energy and lower diffusion kinetic barrier well explain the high separation selectivity of C<sub>2</sub>H<sub>2</sub>/CO<sub>2</sub> in MIL-160.

## CONCLUSIONS

To conclude, we have successfully demonstrated that the high-density hydrogen-bonding nanotrap within the pore surface of porous MOFs can achieve highly selective C<sub>2</sub>H<sub>2</sub>/CO<sub>2</sub> separation. By virtue of the isoreticular principle in MOF chemistry, we can fine-tune the hydrogen-bonding acceptors on the pore surface, as illustrated in three isostructural MOFs. Notably, MIL-160 features the highest density of hydrogen-bonding nanotraps and exhibits a high acetylene storage capacity and superior separation selectivity simultaneously, supplying a new benchmark for C<sub>2</sub>H<sub>2</sub>/CO<sub>2</sub> separation with an excellent separation potential and high C<sub>2</sub>H<sub>2</sub> productivity under ambient conditions. This research provides an outstanding example of a MOF-based hydrogen-bonding nanotrap to address this challenge of gas separation/purification and thus provides a new perspective for rationally designing porous MOF materials in this very active research area.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacs.1c10620>.

Material synthesis, characterization details, and additional figures as described in the text ([PDF](#))

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• Y.Y. and S.X. contributed equally to this work.

## Notes

The authors declare no competing financial interest.

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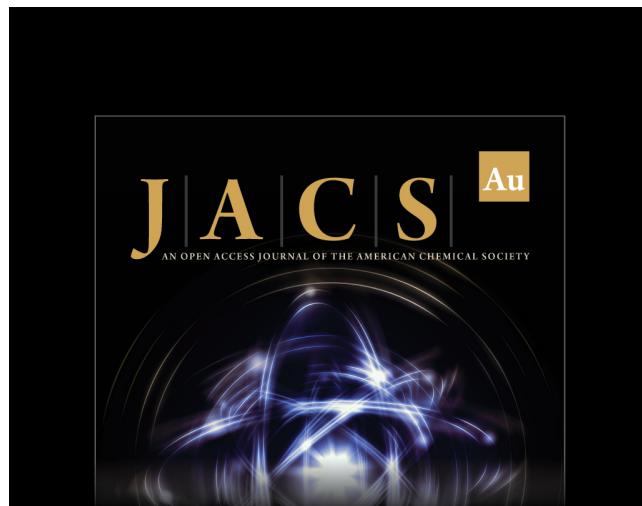
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