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Effect of doping TiO₂ with Mn for electrocatalytic oxidation in acid and alkaline electrolytes†

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Electrochemical oxidation of water and electrolyte ions is a sustainable method for producing energy carriers and valuable chemicals. Among known materials for catalyzing oxidation reactions, titanium dioxide (TiO₂) offers excellent electrochemical stability but is less active than many other metal oxides. Herein, we used density functional theory calculations to predict an increase in catalytic activity by doping anatase TiO₂ with manganese atoms (Mn). We synthesized Mn-doped TiO₂ and then utilized X-ray absorption spectroscopy to study the chemical environment around the Mn site in the TiO₂ crystal structure. Our electrochemical experiments confirmed that TiO₂, with the optimal amount of Mn, reduces the onset potential by 260 mV in a 2 M KHCO₃ (pH = ~8) electrolyte and 370 mV in a 0.5 M H₂SO₄ (pH = ~0.5) electrolyte. Moreover, in 0.5 M H₂SO₄, we observed that the amount of Mn doping greatly impacts the selectivity towards oxygen production versus peroxysulfate formation. In 2 M KHCO₃, the Mn doping of TiO₂ slightly decreases the selectivity towards oxygen production and increases the hydrogen peroxide formation. The Mn-doped TiO₂ shows good electrochemical stability for over 24 hours in both electrolytes.

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Introduction

Electrochemical water splitting can be accelerated with the aid of a catalyst to store and convert renewable energy to hydrogen. The water oxidation reaction (WOR) can proceed through three competing pathways that produce oxygen, hydrogen peroxide, or hydroxyl radicals at the anode of an electrochemical cell.¹ The selectivity towards each product depends on the catalyst material,^{1,2} the electrolyte used,^{3–5} and the potential applied.⁶

The four-electron water oxidation reaction ($4e^-$ WOR, $2H_2O \rightarrow O_2 + 4H^+ + 4e^-$, $E^0 = 1.23$ V),⁷ also known as the oxygen evolution reaction (OER), is the most commonly studied and occurs at the lowest applied potential thermodynamically.⁷ Although the $4e^-$ WOR may not produce a terrestrially valuable product at the anode, it provides the electrons necessary to produce high-purity hydrogen at the cathode. Polymer electrolyte membrane (PEM) electrolysis relies on an acidic electrolyte to operate.^{8,9} While significant efforts aim towards the development of catalysts for the $4e^-$ WOR, severe degradation in acidic conditions has hindered their success.^{10,11}

The two-electron water oxidation reaction ($2e^-$ WOR) requires a larger applied potential ($2e^-$ WOR, $2H_2O \rightarrow H_2O_2 + 2H^+ + 2e^-$, $E^0 = 1.76$ V)⁷ than that of the $4e^-$ WOR but produces valuable hydrogen peroxide (H₂O₂) at the anode. H₂O₂ is a highly sought-after chemical due to its broad applications in industry, home, healthcare, and energy.^{12–14} Unlike PEM electrolysis, the $2e^-$ WOR does not rely on an acidic electrolyte. Rather, an alkaline electrolyte of potassium bicarbonate (KHCO₃) is commonly used.⁵ However, catalyst stability remains a significant challenge for the $2e^-$ WOR due to the higher potential necessary to promote H₂O₂ production.¹

In addition to water oxidation, other reactions can occur *via* oxidation of the electrolyte anions in acidic media. Peroxysulfate (SO₅^{2–} and S₂O₈^{2–}) can be produced at the anode of an electrochemical cell by oxidizing the HSO₄[–] anion in an aqueous H₂SO₄

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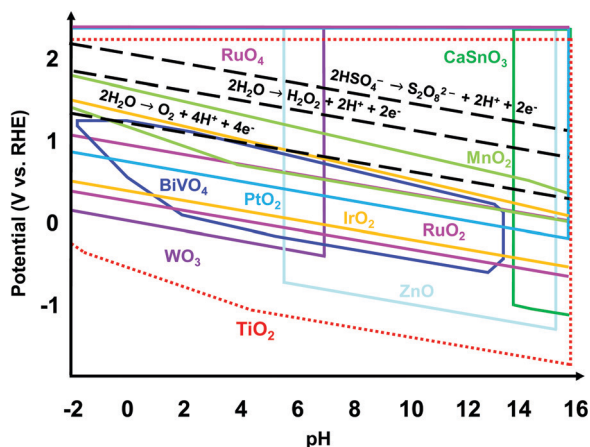


Fig. 1 Overlaid theoretical Pourbaix diagrams of varying metal oxides at ion concentrations of 10 M compiled using Pourbaix diagrams from the Materials Project. Among all those metal oxides, TiO_2 has the broadest stability range in terms of applied potentials and electrolyte pH values, making it an attractive stable electrochemical catalyst.

electrolyte^{15–18} ($2\text{HSO}_4^- \rightarrow \text{S}_2\text{O}_8^{2-} + 2\text{H}^+ + 2\text{e}^-$, $E^0 = 2.08 \text{ V}$).⁷ Similar to H_2O_2 , peroxysulfate has many applications in water purification, chemical synthesis, cleaning, bleaching, and etching.¹⁷ Because $\text{S}_2\text{O}_8^{2-}$ possesses the highest oxidizing capacity among all peroxides,¹⁷ it's considered to be a desirable electrochemical product worth producing. Nevertheless, its catalysts will also face the challenge of retaining stability in acidic conditions and under high anodic bias.

Regardless of the desired electrochemical product, the catalyst used should be electrically stable under applied potential and chemically stable in the acidic or alkaline conditions of the electrolyte. Fig. 1 shows the theoretically stable pH and potential ranges of several metal oxides used as electrocatalysts. The Pourbaix diagrams were compiled from the Materials Project database and assume ion concentrations of 10 M.^{19,20} Black dashed lines were added to indicate the applied potentials necessary to produce O_2 , H_2O_2 , and $\text{S}_2\text{O}_8^{2-}$. The stable regions of many metal oxides do not reach or barely reach these redox lines, indicating that they would not be stable

at the applied potentials necessary to produce these products. Ru and Ir and their oxides are the most well-known and some of the top-performing catalysts for the 4e^- WOR to O_2 in acidic conditions.^{10,21} However, RuO_x and IrO_x are susceptible to further oxidation and dissolution under applied voltage.²² Additionally, Ru and Ir are scarce and expensive. Doped BiVO_4 has been shown to have high selectivity towards the 2e^- WOR pathway to H_2O_2 but suffers from severe degradation.²³ WO_3 has proven to produce peroxysulfate in an aqueous H_2SO_4 electrolyte but only under solar irradiation.^{15–18} Among all the metal oxides shown, TiO_2 has the widest stable range of pH values and applied potentials, making it a great candidate as a stable electrochemical catalyst. In addition, TiO_2 is a nontoxic, earth-abundant, and low-cost material.^{24,25} That being said, TiO_2 alone requires a high overpotential for water oxidation.^{6,26}

Heterometal electrocatalysts have become a growing research interest in recent years due to their tunable activity and selectivity and the improvement of theoretical modeling and characterization techniques used to study these materials at the atomic level.²⁷ In particular, various manganese (Mn)-doped TiO_2 structures have been synthesized for a wide range of catalytic applications, such as nanosheets,²⁸ nanopowders,^{29,30} nanoparticles,^{31–33} and thin films³⁴ as photocatalysts, nanospheres for electrochemical N_2 reduction,³⁵ nanoparticles for electrochemical O_2 reduction,³⁶ and thin films for water oxidation.³⁷ Herein, we study the effect Mn atoms have on the activity and selectivity of TiO_2 for different electrochemical oxidation pathways in 0.5 M H_2SO_4 and 2 M KHCO_3 electrolytes. We synthesized Mn-doped TiO_2 ($\text{Mn}:\text{TiO}_2$) thin films *via* a simple and rapid sol-gel method and verified the nature of Mn through X-ray Absorption Spectroscopy (XAS). Our density functional theory (DFT) calculations and experiments show that a moderate amount of Mn improves the onset potential in both acid and alkaline electrolytes while retaining TiO_2 's superior stability.

Theoretical analysis

We first performed DFT calculations to study the 4e^- WOR activity over the $\text{Mn}:\text{TiO}_2$ systems. Optimized structures of bulk anatase TiO_2 were cleaved to the (100), (101), and (001)

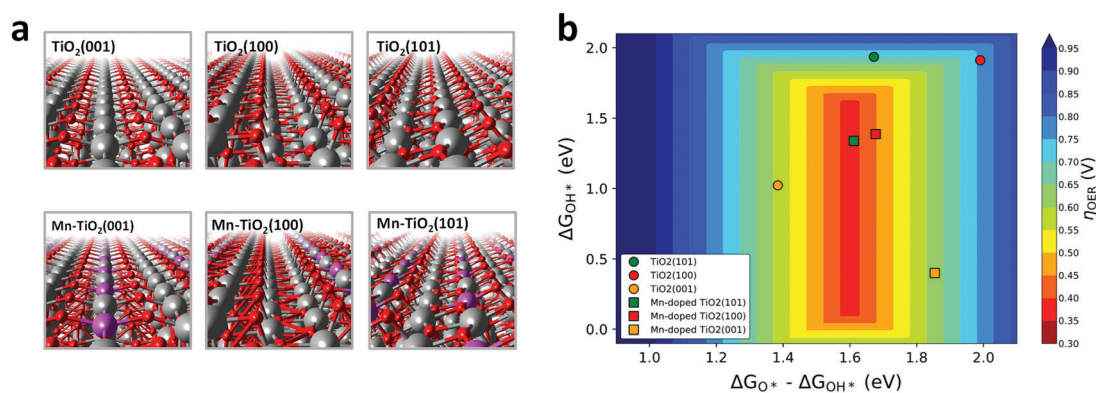
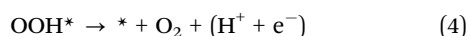
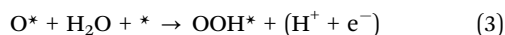
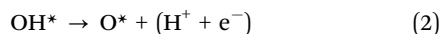
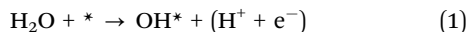


Fig. 2 (a) Side view of the optimized TiO_2 and $\text{Mn}:\text{TiO}_2$ surfaces studied *via* DFT. Grey, purple, and red spheres represent Ti, Mn, and O atoms, respectively. (b) 4e^- water oxidation reaction activity volcano plot showing the calculated overpotential, η_{OER} , as a function of adsorption free energies, ΔG , of reaction intermediates.

low-index facets, which are most commonly exposed on anatase TiO₂.^{38,39} Single atoms of Mn were introduced into the surface layer of the TiO₂ lattices to explore its effect on the WOR (Fig. 2a). We also considered subsurface doping and a fully Mn-saturated TiO₂ surface (Fig. S1, ESI†). All resulting structures were used to model the 4e[−] WOR reaction mechanism described by eqn (1)–(4), where * denotes a surface Mn or Ti active site.



The intrinsic electrocatalytic activity was examined based on thermodynamics, which is a commonly employed method used to rationalize trends in various electrochemical reactions involving oxygen.^{13,23,40} We apply the computational hydrogen electrode (CHE) model in which the chemical potential of an electron–proton couple (H⁺ + e[−]) is equivalent to that of the gas phase H₂ at *U* = 0.0 V vs. reversible hydrogen electrode (RHE).⁴¹ Through the CHE model, the Gibbs free energy of adsorption, Δ*G*, of reaction intermediates on the surface of the Mn:TiO₂ were calculated (see ESI† for calculation details).⁴¹ The calculated limiting potential (*U*_L) is defined as the minimum potential at which all steps are downhill in free energy. From which, theoretical overpotential, η, can be calculated.

$$\eta_{\text{Theor.}}^{\text{OER}} = U_{\text{L}} - 1.23 \text{ eV} \quad (5)$$

Fig. 2b displays the theoretical 4e[−] WOR overpotential for the doped and undoped TiO₂ as a function of Δ*G*_{OH*} and Δ*G*_{O*−}

Δ*G*_{OH*} in a 3D volcano contour plot. Δ*G*_{OH*} and Δ*G*_{O*−}Δ*G*_{OH*} are widely reported as effective descriptors for the activity towards the WOR.^{42,43} This is in part due to a proposed “universal” linear scaling relation between Δ*G*_{OOH*} and Δ*G*_{OH*} used to describe a variety of metal oxides, Δ*G*_{OOH*} = Δ*G*_{OH*} + 3.2 eV (±0.2 eV).^{42–44} Ideally, the catalyst will lie in the red region at the center of the plot, with optimal intermediate binding free energies, minimizing overpotentials. TiO₂ (sphere symbols, in Fig. 2b) sits at the top, right-hand side of the volcano plot where the OH* intermediate is weakly bound to a surface Ti atom. With the introduction of a surface Mn into the TiO₂ lattice (square symbols, Fig. 2b), the interaction with reaction intermediates becomes stronger, shifting the Δ*G*_{OH*} and Δ*G*_{O*−}Δ*G*_{OH*} descriptors to lower values. Consequently, Mn:TiO₂ surfaces have lower calculated overpotentials than those of TiO₂ alone as Mn strengthens the adsorption of all reaction intermediates and shifts it towards optimal binding energies.

To gain insights into the nature of the Mn–oxygen intermediate interaction, a Bader charge analysis and charge density difference plots were carried out (Fig. S2–S3, ESI†). An in-depth discussion can be found in the ESI.† This analysis revealed a greater charge transfer between the surface Mn atom (in Mn:TiO₂) and OH*, O*, or OOH* adsorbates when compared to that between a surface Ti atom (in TiO₂) and adsorbates. As binding events are often accompanied by a redistribution of charge, a greater charge transfer suggests a preference in Mn–oxygen intermediate binding, relative to that in the Ti active site. Accordingly, our results indicate that the decrease in calculated overpotentials, observed upon the incorporation of Mn into the TiO₂ surface (Fig. 2b), originates from a greater charge transfer between the oxygenated reaction intermediate and the active site (*i.e.*, Mn or Ti).

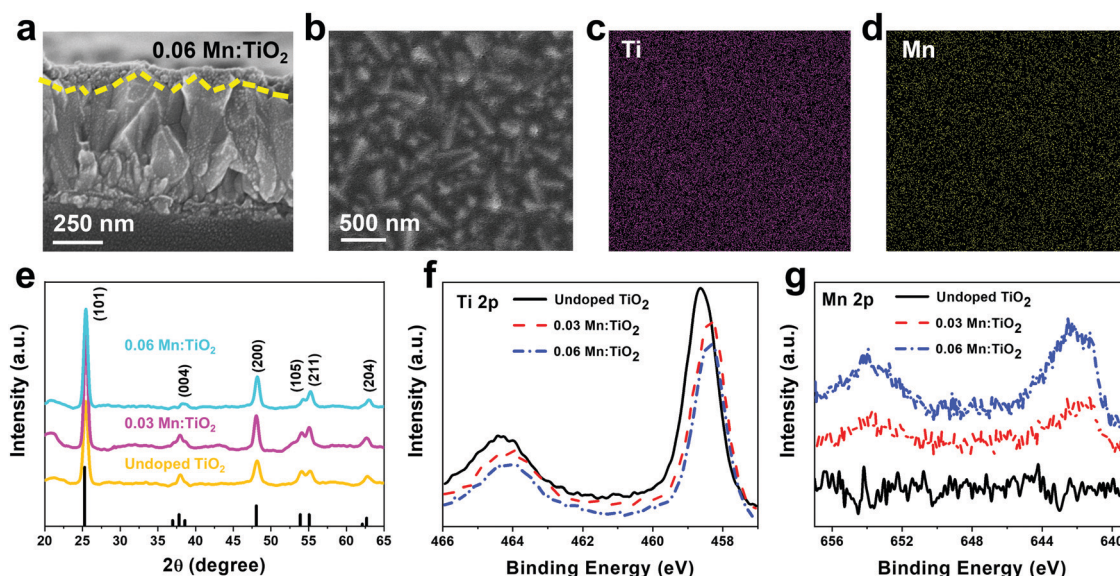


Fig. 3 (a) Side-view SEM image of 0.06 Mn:TiO₂ on FTO glass showing a sample average width of ~100 nm. (b) SEM-EDS elemental mapping of (c) Ti and (d) Mn in 0.06 Mn:TiO₂. (e) XRD spectra of undoped TiO₂, 0.03 Mn:TiO₂, and 0.06 Mn:TiO₂. Anatase TiO₂ reference peaks are shown in black from ICDD: No. 21-1272. XPS spectra of (f) Ti 2p and (g) Mn 2p peaks confirm both Ti and Mn are prevalent in the 0.03 and 0.06 Mn:TiO₂ samples and no Mn is prevalent in the undoped sample.

Material characterization

Thin films of Mn:TiO₂ with Mn to Ti atomic ratios of 0 (undoped), 0.03, and 0.06 were synthesized on fluorine-doped tin oxide (FTO) coated glass using a sol-gel-method. As for notation, we use X Mn:TiO₂, where X represents the atomic ratio of Mn to Ti in the sol-gel precursor solution. The cross-section scanning electron microscope (SEM) image in Fig. 3a shows that the film of 0.06 Mn:TiO₂ has an average thickness of ~100 nm. An aerial SEM image (Fig. 3b) was used for energy-dispersive X-ray spectroscopy (EDS) of Ti (Fig. 3c) and Mn (Fig. 3d) showing homogeneous distribution. The crystal structure was investigated by X-ray diffraction (XRD) and was determined to be anatase TiO₂ for the undoped, 0.03, and 0.06 Mn:TiO₂ samples, as shown in Fig. 3e, where the black peaks correspond to reference anatase TiO₂ (ICDD: No. 21-1272). The average crystallite size of TiO₂ was estimated to be approximately 10 nm based on the Scherrer equation. If the Mn to Ti atomic ratio is extended past 0.06 to 0.09 or 0.12, the TiO₂ transitions from anatase to rutile phase according to XRD spectra (Fig. S4, ESI[†]).⁴⁵ The chemical state and composition of the surface of the samples were analyzed using X-ray photoelectron spectroscopy (XPS). Fig. 3f shows the Ti 2p XPS spectra for all samples. The Ti 2p peaks of the undoped sample occur at 464.4 eV and 458.6 eV, which we ascribed to the published Ti 2p_{1/2} and 2p_{3/2} binding energies of Ti⁴⁺ in TiO₂.^{46,47} For the 0.03 and 0.06 Mn:TiO₂ samples, both Ti 2p peaks are slightly shifted to lower binding energies by 0.3 eV, likely due to the weakening of the Ti–O bonds from the incorporation of Mn in the TiO₂ lattice.⁴⁸ The Mn 2p XPS spectra in Fig. 3g show that Mn is present only in the doped samples. The bulk XRD characterization confirms that the samples are crystalline anatase TiO₂ without Mn-based impurity phases, while the surface XPS characterization identifies Mn in the doped anatase TiO₂ structures.

To determine the oxidation state of the Mn atoms and gain insights into the local atomic structure of Mn in the TiO₂ lattice, the Mn:TiO₂ samples were studied using X-ray absorption spectroscopy (XAS). The reference spectra of MnO, Mn₃O₄, and MnO₂ standards were also obtained for comparison purposes. Fig. 4a shows the normalized Mn K-edge X-ray absorption near edge structure (XANES) spectra of the standards and the 0.03 and 0.06 Mn:TiO₂ samples. The Mn K-edge XANES of the doped samples shows distinct spectral features and a different absorption edge energy position (the first maximum in the first derivative) than the oxide standards. The 0.03 and 0.06 Mn:TiO₂ doped samples exhibited mostly overlapped spectra, suggesting that Mn is in a similar micro-environment. The pre-edge peak centered at around 6540 and 6542 eV appears to be composed of a double component (based on the broadness of the feature), which may be characteristic of an octahedral environment (inset in Fig. 4a).⁴⁹ A shift toward higher energy has been observed in this pre-edge peak position with an increasing valence state.⁵⁰ In general, as the oxidation state increases, the energy at which the main absorption peak occurs increases as well.⁵¹ This trend is clearly shown in Fig. 4b by comparing the energy of the absorption edge to the

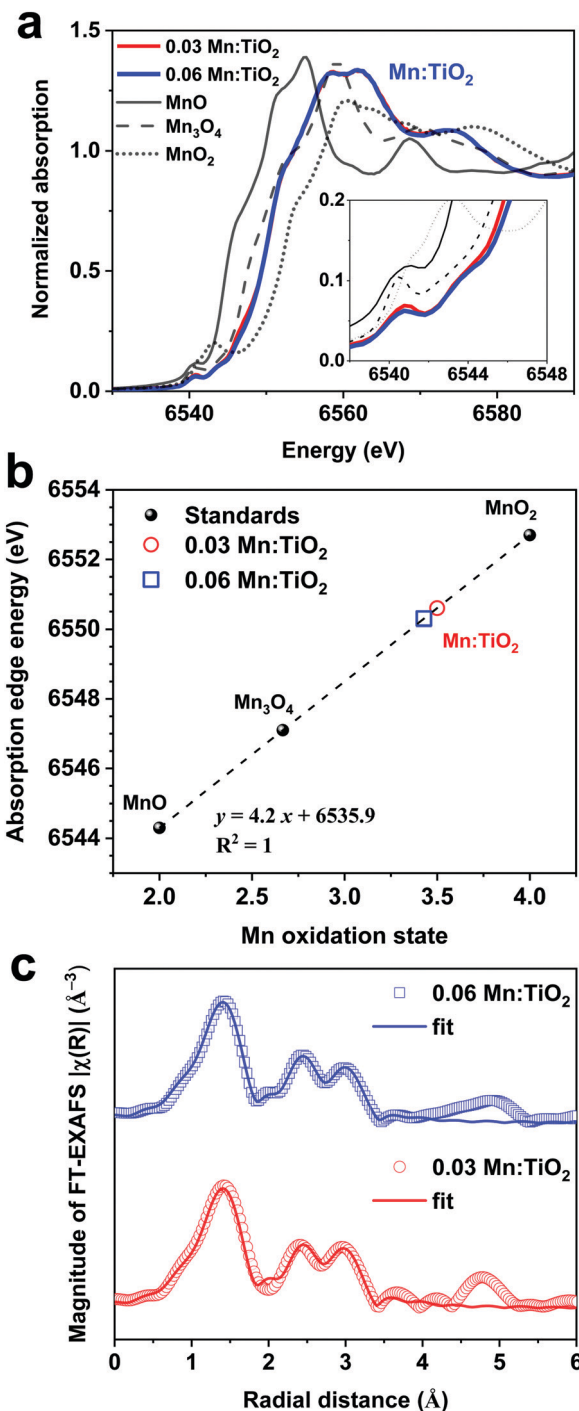


Fig. 4 (a) Mn K-edge X-ray Absorption near-edge structure (XANES) spectra of 0.03 and 0.06 Mn:TiO₂ and references MnO, Mn₃O₄, and MnO₂. The inset highlights the XANES pre-edge peaks. (b) Comparing the XANES absorption edge energies of the standards MnO, Mn₃O₄, and MnO₂ to estimate the oxidation states of the Mn:TiO₂ samples. Using this method, the Mn oxidation state of the Mn:TiO₂ samples was determined to be ~3.5. (c) Mn K-edge Fourier-transformed extended X-ray absorption fine structure (FT-EXAFS) distribution functions for the 0.03 and 0.06 Mn:TiO₂ samples without phase correction along with the obtained theoretical fits from the FT-EXAFS analysis. The symbols show the experimental data, and the solid lines are the theoretical fittings.

oxidation states of manganese(II) oxide (MnO), manganese(II,III) oxide (Mn_3O_4), and manganese(IV) oxide (MnO_2). Overall, an oxidation state of 3.5 was obtained for the Mn:TiO₂ samples. It has been reported that the Mn K-edge position can be affected by local distortions of the MnO_6 octahedra;⁵² nonetheless, when also considering a complementary first-moment analysis (Fig. S5, ESI[†]), our spectroscopy results indicate that the average oxidation state of the Mn sites in the doped anatase sample is between 3.1 and 3.5.

Extended X-ray absorption fine structure (EXAFS) was performed to study the average local atomic environment of the Mn atom that is assumed to substitute a Ti atom in the anatase crystal structure. Fig. 4c shows the Mn K-edge Fourier-transformed EXAFS (FT-EXAFS) analysis of the 0.03 and 0.06 Mn:TiO₂ samples. As an initial step, a first-shell fit indicates that the Mn atom coordinates with approximately 5.8 ± 0.4 O-atoms at a distance of 1.91 ± 0.01 Å, most likely in an octahedral environment (Fig. S6 and Table S1, ESI[†]). Second, the long-range fitting indicates that the Mn–Ti coordination number (CN) of the 0.03 and 0.06 samples can vary from 1.1 to 1.8 when assuming Mn substitutions in configurations such as on the (101) and (001) surfaces or in the bulk anatase lattice (average $\text{CN}_{\text{Mn-Ti}} = 1.4 \pm 0.7$ at an average distance of 2.89 ± 0.02 Å). Third, the Mn–O–Ti multiple scattering paths and long-

range Mn–O' scattering path resulted in an average $\text{CN}_{\text{Mn-O-Ti}}$ of 5 ± 2 ($R_{\text{Mn-O-Ti}} = 3.77 \pm 0.02$ Å) and $\text{CN}_{\text{Mn-O'}}$ of 16 ± 2 ($R_{\text{Mn-O'}} = 3.97 \pm 0.03$ Å), respectively (see details of the FT-EXAFS fitting in Fig. S7 and Tables S2–S5, ESI[†]). Those longer scattering path's coordination numbers are in reasonable agreement with the expected anatase values of 4 (Mn–O–Ti) and 16 (Mn–O').

Ultimately, the XANES and EXAFS analysis indicate that the Mn atom has an average oxidation state of 3.3 in an octahedral environment with an average of 5.8 O-atoms at 1.9 Å. The best theoretical fit indicates an average number of 1.5 Ti-atoms at a 2.9 Å distance from the Mn center, which is lower than the expected value of 4 for bulk anatase. The Mn K-edge EXAFS provides an average perspective of the local coordination environment of Mn atoms in the sample, and although it cannot rule out Mn–Mn coordination, the extracted low Mn–Mn CN (*i.e.*, Mn to Ti or Mn to Mn with ≈ 2.9 Å distance) may support the proposal that Mn sites are homogeneously dispersed mostly on anatase surfaces. Based on the doping level and assuming spherical 10 nm TiO₂ nanoparticles (estimated by XRD), we estimate a surface-to bulk atomic ratio where approximately one-third of the Ti atoms in the system are surface atoms with a plausible percentage of Mn substitutions. DFT suggests (101) surfaces to be preferentially exposed

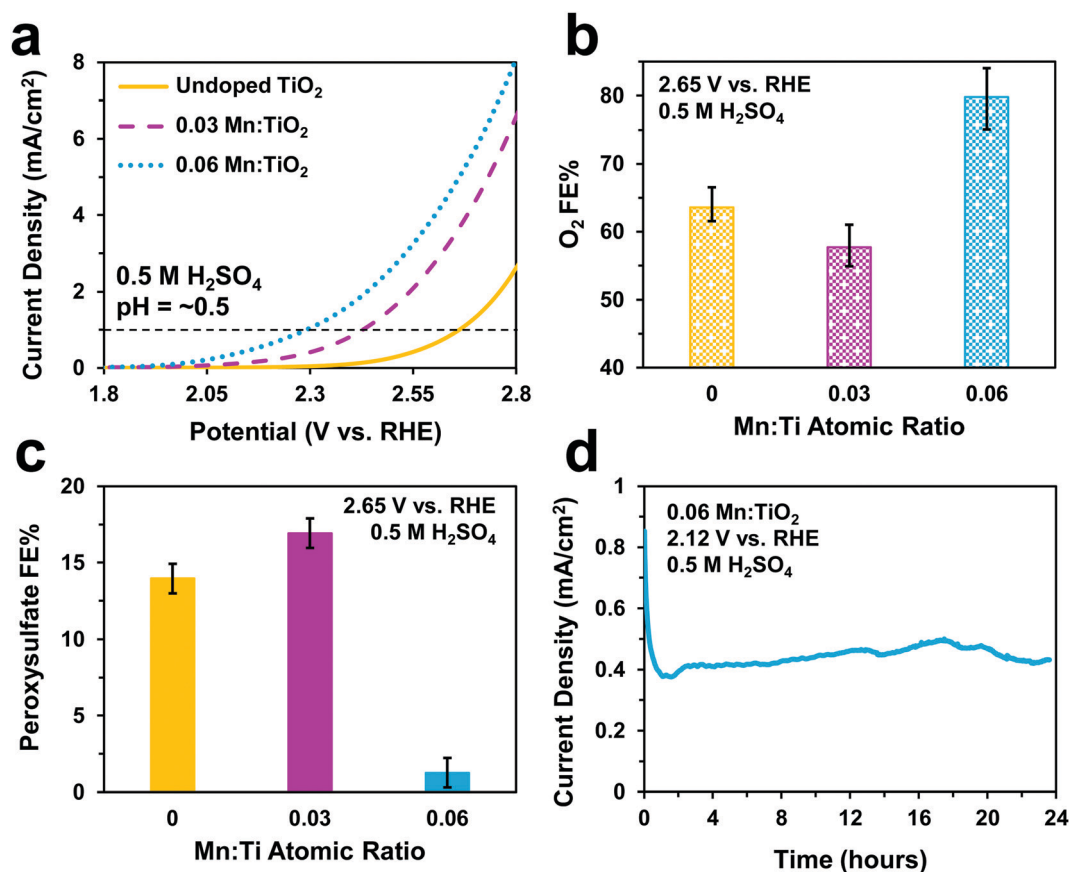


Fig. 5 Electrochemical performance in a 0.5 M H₂SO₄ electrolyte. (a) Linear sweep voltammograms of undoped TiO₂, 0.03 Mn:TiO₂, and 0.06 Mn:TiO₂ at a scan rate of 50 mV s^{−1}. (b) Faradaic efficiencies toward O₂ under an applied potential of 2.65 V vs. RHE. (c) Faradaic efficiencies toward peroxysulfate after an applied potential of 2.65 V vs. RHE for 10 minutes. (d) Stability test of 0.06 Mn:TiO₂ at 2.12 V vs. RHE.

(i.e., lowest surface energy); yet, on 10 nm TiO₂ nanoparticles, vacancies and other surface defects are abundant exposing a higher density of unsaturated Mn-substituted Ti sites. Notably, the extracted lower Mn to Ti coordination suggests that Mn sits isolated preferentially on the surface.

Electrocatalytic oxidation experiments

Electrochemical tests were performed to evaluate the catalytic performance of undoped TiO₂, 0.03, and 0.06 Mn:TiO₂ samples for O₂ and S₂O₈²⁻ production in 0.5 M H₂SO₄ (pH = ~0.5). As predicted, doping TiO₂ with Mn lowers the onset potential as shown in Fig. 5a (with an onset potential defined as the potential applied to achieve 1 mA cm⁻²). The onset potential of 0.06 Mn:TiO₂ is 370 mV lower than that of the undoped sample. The faradaic efficiency (FE) towards O₂ (Fig. 5b) and S₂O₈²⁻ (Fig. 5c) were evaluated after 5 minutes at 2.65 V vs. RHE, and the amount of Mn dopant was found to have a significant impact on the FE. For the 0.03 Mn:TiO₂ sample, 58% of the charge goes towards producing O₂ and 17% towards S₂O₈²⁻, while for the 0.06 Mn:TiO₂ sample, 80% of the charge goes towards producing O₂ and only 1% towards S₂O₈²⁻. Lastly, the stability of the 0.06 Mn:TiO₂ sample in the 0.5 M H₂SO₄ electrolyte was evaluated at 2.12 V vs. RHE. After the decay in current density due to charging of the electrochemical double

layer was considered, the current density change is within 5% over the 24 hours tested (Fig. 5d).

Next, we studied the electrocatalytic performance of the samples in 2 M KHCO₃ (pH = ~8), an electrolyte known to aid in the production of H₂O₂.⁵ In Fig. 6a, we show a similar activity trend as observed in 0.5 M H₂SO₄, where an increase in Mn lowers the onset potential. The 0.06 Mn:TiO₂ decreases the onset potential by 260 mV compared to the undoped sample. As for selectivity, increasing the Mn amount in Mn:TiO₂ decreases the FE towards O₂ from 87% to 75% after applying 3 V vs. RHE for 5 minutes (Fig. 6b), opposite the trend observed in the acid electrolyte (Fig. 5b), but it increases the FE towards H₂O₂ (Fig. 6c). Here, we noticed that the FE towards H₂O₂ shows a dependence on the length of time the potential was applied (5, 20, or 40 minutes), where the H₂O₂ FE decreases with increasing testing time. For example, for the 0.06 Mn:TiO₂ sample, the FE towards H₂O₂ decreases from 14% after 5 mins of testing to 7% after 40 mins of testing. Although the exact reasoning for this time dependence is unknown, one possibility is that the produced H₂O₂ is further oxidized at the surface of the Mn:TiO₂ catalyst, resulting in a lower H₂O₂ FE after a longer testing time. Another possibility is that the H₂O₂ could be oxidizing the carbonate ions in the electrolyte during the longer timescales, in which case the H₂O₂ would get converted back to

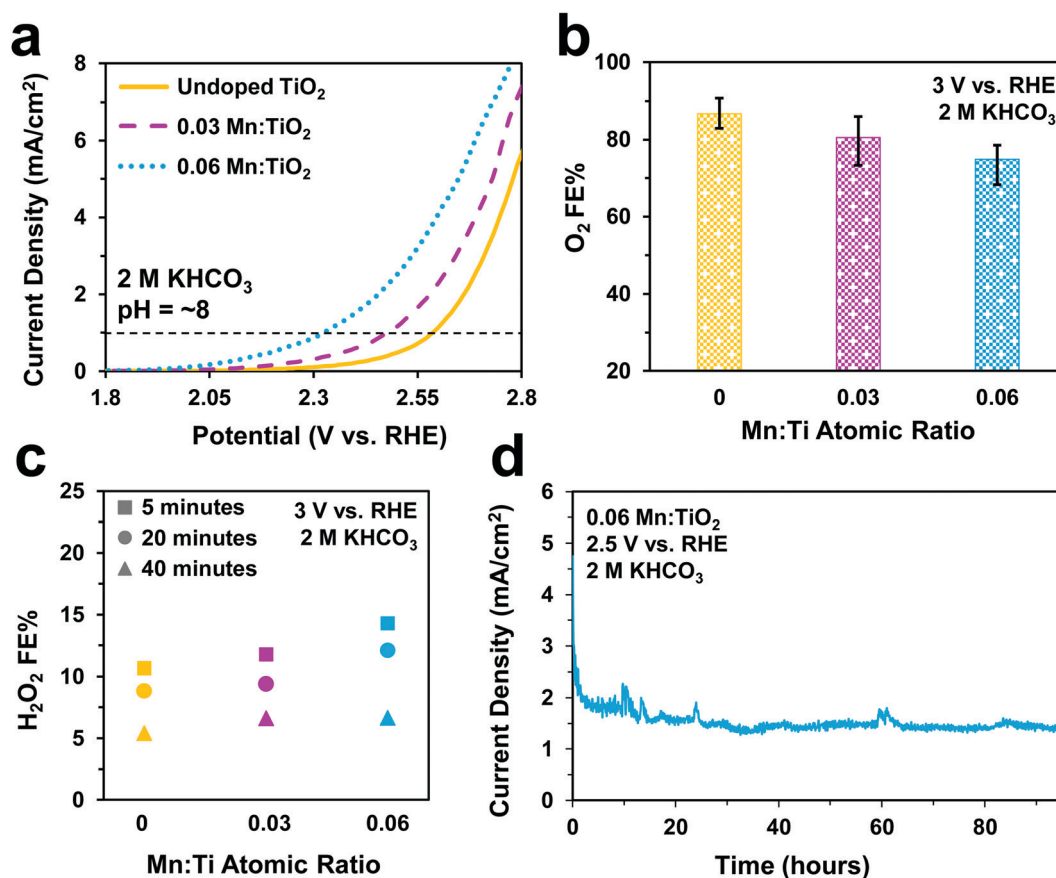


Fig. 6 Electrochemical performance in a 2 M KHCO₃ electrolyte. (a) Linear sweep voltammograms of undoped TiO₂, 0.03 Mn:TiO₂, and 0.06 Mn:TiO₂ at a scan rate of 50 mV s⁻¹. (b) Faradaic efficiencies toward O₂ under an applied potential of 3 V vs. RHE. (c) Faradaic efficiencies towards H₂O₂ after an applied potential of 3 V vs. RHE for 5, 20, and 40 minutes. (d) Stability test of 0.06 Mn:TiO₂ at 2.5 V vs. RHE.

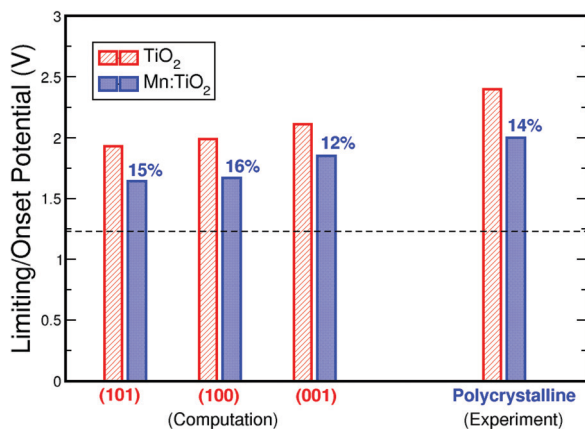


Fig. 7 Comparison of the calculated and measured onset potentials for undoped TiO₂ and Mn:TiO₂, showing an agreement between computational theory and experiments. The dashed line represents the thermodynamic limit of water oxidation for O₂ production at 1.23 V vs RHE.

water. The stability of the 0.06 Mn:TiO₂ sample in 2 M KHCO₃ was evaluated under 2.5 V vs. RHE. It retained ~93% of its original current density in the first 24 hours and ~69% of its original current density after four days (Fig. 6d), after taking into account the initial decay in current density due to charging of the electrochemical double layer.

Finally, Fig. 7 compares the theoretical and experimental onset potentials of Mn-doped TiO₂ and undoped TiO₂. Here, we show the calculated onset potentials assuming exposed facets of (101), (100), and (001) on anatase TiO₂. The experimentally measured onset potentials were taken from the current density–voltage curves of undoped TiO₂ and 0.06 Mn:TiO₂ in 0.5 M H₂SO₄. As predicted by theory, a substantial improvement is observed experimentally from the addition of Mn in the TiO₂ lattice. When Mn substitutes Ti, theoretically, the onset potential decreases by 15%, 16%, and 12% for (101), (100), and (001) respectively. Experimentally, the onset potential decreases by 14%, which is in excellent agreement with theory.

Conclusions

The catalytic activity of TiO₂ was improved by homogeneous dispersion of Mn sites on the surface of anatase TiO₂, as predicted by DFT calculations. In the 0.06 Mn:TiO₂ sample, the onset potential was reduced by 370 mV, and the sample was proven to be stable in 0.5 M H₂SO₄ at 2.12 V vs. RHE for 24 hours. Applying a voltage of 2.65 V vs. RHE to the 0.03 Mn:TiO₂ sample resulted in a FE towards S₂O₈^{2−} of 17%. In a 2 M KHCO₃ electrolyte, the onset potential was reduced by 260 mV, and the sample was proven to be stable at 2.5 V vs. RHE for 4 days with only a 7% loss in current density after the first 24 hours. These experiments confirm TiO₂'s superior stability in highly oxidizing conditions from extreme pH values and potentials. In the future, the onset potentials need to be further decreased for the OER, and the FEs towards H₂O₂ and S₂O₈^{2−} could be increased by potentially incorporating other metal dopants on well-defined selective facets. The current density needs to be

increased, which could be achieved by adding small amounts of an additional active and highly conductive dopant. Synthesis of Mn:TiO₂ could also be optimized for light absorption in hopes of lowering applied overpotentials *via* photoelectrochemistry.

Methods

Computational details

All electronic structure calculations were performed using the QUANTUM ESPRESSO program package.⁵³ Atomic Simulation Environment (ASE) was used to handle the simulation.⁵⁴ The electronic wavefunctions were expanded in plane waves up to a cut-off energy of 500 eV, while the electron density is represented on a grid with an energy cut-off of 5000 eV. Core electrons were approximated with ultrasoft pseudopotentials.⁵⁴ The Perdew–Burke–Ernzerhof (PBE) functional was used to describe the chemisorption properties. TiO₂ and Mn:TiO₂ surfaces were modeled using a 2 × 2 unit cell with a total of four layers for the (101), (001), and (100) facets. The first two layers and adsorbates were allowed to relax while the two bottom layers were fixed at their bulk position. A vacuum of ~20 Å was used to decouple periodic replicas along the z-direction. The Brillouin zone was sampled with (4 × 4 × 1) Monkhorst–Pack *k*-points.

Material synthesis

Sheets of 7–8 Ohm per square TEC 7 FTO coated glass (MSE Supplies) were cut and sonicated in acetone, isopropyl alcohol, and DI water for 10 minutes each before being rinsed in DI water and dried. TiO₂ was synthesized on the FTO substrates *via* a solution gelation (sol–gel) method.⁵⁵ To prepare a solution for dip coating, 38.5 mL of ethanol was placed in a glass container in an ice bath and stirred while 0.0385 mol (11.42 mL) of titanium(IV) isopropoxide (TTIP) 97% (Sigma-Aldrich) was added dropwise. Next, a second solution was prepared of 50 mL of ethanol, 2.5 mmol of HCl, and 30 mmol of deionized water, which was stirred and incrementally added to the first solution of ethanol and TTIP. The combined ~100 mL solution was sonicated for 10 minutes in an ice bath. The solution was then stirred for 4.5 hours at room temperature. The solution should remain transparent. For the Mn-doped samples, manganese(II) acetate 98% (Sigma-Aldrich) was added after stirring, and the solution was sonicated for another 15 minutes in an ice bath. For a 0.03 Mn to Ti atomic ratio, 0.0012 mol of Mn was added. For a 0.06 Mn to Ti atomic ratio, 0.0023 mol of Mn was added. After sonication, substrates of FTO coated glass were dip-coated into the solution while the solution was being stirred in an ice bath. A section of the FTO was kept uncoated for electrical contact. The samples were then put in a furnace at 260 °C for 10 minutes to promote gelation. After cooling, the samples were annealed at 500 °C for 2 hours. Copper tape was attached to the exposed FTO, and silver paste was painted over the connection. Any exposed FTO was isolated with glue so that only the dip-coated TiO₂ or Mn:TiO₂ was exposed to the electrolyte during testing. This exposed area was measured using ImageJ software and was used as the geometric area to normalize the current densities.



Material characterization

The surface morphology and sample thickness were viewed using scanning electron microscopy (FEI XL30, Sirion). The crystallinity of TiO₂ was investigated by X-ray diffraction (Xpert2, PANalytical with Cu K α radiation) deposited on silicon dioxide substrates to avoid measuring peaks from FTO. The chemical state on the surface was determined using X-ray photoelectron spectroscopy (PHI Versaprobe 1 and Versaprobe 3 with Al K α radiation) calibrated to adventitious carbon at 248.8 eV. Mn K-edge X-ray absorption spectroscopy was collected at BL 7-3 at the Stanford Synchrotron Radiation Light source (SSRL) with a Si (220) monochromator 50% detuned that delivered a 1 mm(v) $\times$ 3 mm(h) beam. XANES and EXAFS were done in fluorescence mode directly on the FTO-supported electrodes. Reference spectra were simultaneously collected upstream using Mn metal *via* transmission detection mode, and all scans were calibrated to the metallic reference at 6539 eV. At least four scans were averaged, and a linear pre-edge was subtracted. Athena software was used for data normalization by unit edge jump. For the EXAFS analysis, multiple *k*-weighted EXAFS data in *R* space were obtained by subtracting a polynomial background function to the normalized data and processing the resulting $\chi(k)$ signal through Fourier transform (using a Hanning window, $\Delta k = 1$, *k*-range of 3–10.5 Å^{−1}, *R*-range of 1.0–1.8 for the first shell, and *R*-range of 1.0–3.7 for the long-range fitting). Data were fitted in *R*-space using the Artemis software.⁵⁶

Electrochemical testing

Electrochemical tests were run using a Gamry 1000 potentiostat with a three-electrode configuration. Synthesized undoped TiO₂ or Mn-doped TiO₂ on FTO coated glass was used as the working electrode for all experiments. When testing in an electrolyte of 0.5 M H₂SO₄, a Pt wire was used as the counter electrode and Hg/Hg₂SO₄ (Gamry) as the reference electrode. When testing in an electrolyte of 2 M KHCO₃, a piece of carbon paper (Fuel Cell Store) or a graphite rod (Gamry) was used as the counter electrode and Ag/AgCl (Gamry) as the reference electrode. To condition each sample, the voltage was cycled 10 times between 0 V *vs.* ref. and the voltage applied during subsequent chronoamperometry tests. The voltage was linearly increased in increments of 10 mV at 50 mV s^{−1} before a constant voltage was applied for ≥ 5 minutes to produce a measurable quantity of product or to evaluate the catalyst's long-term stability. The electrolyte was constantly stirred during the chronoamperometry tests using a magnetic stir bar to prevent concentration gradients.

Oxygen quantification

The amount of dissolved oxygen was measured both in 0.5 M H₂SO₄ and 2 M KHCO₃ electrolytes using a HIOXY-R oxygen probe attached to an oxygen sensor (NeoFox, Ocean Optics). The electrodes, oxygen probe, gas bubbler, and electrolyte were contained within a sealed cell (EuroCell, Gamry). The oxygen probe and gas bubbler were placed mid-depth into the

electrolyte. The oxygen sensor was calibrated against an air-saturated and argon-saturated electrolyte. Before each experiment, the stirred electrolyte was purged with argon *via* the gas bubbler until the oxygen measurement was stable. The bubbler was then removed from the cell and the port was closed to ensure an airtight seal. The oxygen concentration was continuously measured throughout the chronoamperometry test, and the FE was calculated using the following equation:

$$FE_{O_2} = \frac{n \times [O_2]_{mol} \times V \times F}{q} \times 100 \quad (6)$$

where FE_{O_2} is the faradaic efficiency toward O₂ expressed as a percentage, *n* is the stoichiometric number of electrons transferred for O₂ production *via* water oxidation (*n* = 4), [O₂] is the concentration of accumulated O₂ in mol L^{−1}, *V* is the electrolyte volume in L, *F* is Faraday's constant, and *q* is the charge passed in coulombs.

The accumulated moles of O₂ in the electrolyte increased steadily while the voltage was applied until shortly after the experiment ended when the O₂ concentration plateaued. The concentration never approached the O₂ solubility limit of 1.22 mmol L^{−1}.⁵⁷ The O₂ concentration in the headspace above the electrolyte was also measured before and after the chronoamperometry test to ensure a negligible change.

Peroxydisulfate and hydrogen peroxide quantification

The amount of S₂O₈^{2−} generated in 0.5 M H₂SO₄ was measured after a voltage of 2.65 V *vs.* RHE was applied for 10 minutes. A 2 mL aliquot was taken from the electrolyte, and 2 mL of 0.01 M FeSO₄ and 2 mL of 1 M H₂SO₄ were then added to the aliquot.¹⁸ The absorbance of Fe(III) was measured using UV-visible spectrophotometry (Agilent Cary 6000i) at 302 nm.⁵⁸ A calibration was performed using various concentrations of K₂S₂O₈ sonicated in 0.5 M H₂SO₄.

The amount of H₂O₂ generated in 2 M KHCO₃ was measured after a voltage of 3 V *vs.* RHE was applied for either 5, 20, or 40 minutes. An aliquot of the electrolyte was mixed with 2 M KHCO₃ and 0.07 M cobalt sulfate heptahydrate, forming a complex of Co(CO)₃^{3−}.⁵⁹ The absorbance of the cobalt-carbonate assay was measured using UV-visible spectrophotometry at 260 nm.⁶⁰ The FEs toward S₂O₈^{2−} and H₂O₂ were calculated using the following equation:

$$FE_{H_2O_2/S_2O_8} = \frac{n \times [H_2O_2/S_2O_8^{2-}] \times V \times F}{q \times M} \times 100 \quad (7)$$

where $FE_{H_2O_2/S_2O_8}$ is the faradaic efficiency toward H₂O₂ or S₂O₈^{2−} expressed as a percentage, *n* is the stoichiometric number of electrons transferred for H₂O₂ or S₂O₈^{2−} production (*n* = 2), [H₂O₂/S₂O₈^{2−}] is the concentration of H₂O₂ or S₂O₈^{2−} in g mL^{−1}, *V* is the electrolyte volume in mL, *F* is Faraday's constant, *q* is the charge passed in coulombs, and *M* is the molar mass of H₂O₂ or S₂O₈^{2−} in g mol^{−1}.

Nomenclature

WOR	Water oxidation reaction
OER	Oxygen evolution reaction



PEM	Polymer electrolyte membrane
Mn:TiO ₂	Mn-doped TiO ₂
DFT	Density functional theory
*	Surface active site
CHE	Computational hydrogen electrode
RHE	Reversible hydrogen electrode
ΔG	Gibbs free energy of adsorption
η	Overpotential
FTO	Fluorine-doped tin oxide
SEM	Scanning electron microscopy
EDS	Energy-dispersive X-ray spectroscopy
XRD	X-Ray diffraction
XPS	X-Ray photoelectron spectroscopy
XAS	X-Ray absorption spectroscopy
XANES	X-Ray absorption near edge structure
EXAFS	Extended X-ray absorption fine structure
FT-EXAFS	Fourier transformed extended X-ray absorption fine structure
CN	Coordination number
FE	Faradaic efficiency

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References

- 1 X. Shi, S. Back, T. M. Gill, S. Siahrostami and X. Zheng, *Chem*, 2021, **7**(1), 38–63.
- 2 C. C. L. McCrory, S. Jung, I. M. Ferrer, S. M. Chatman, J. C. Peters and T. F. Jaramillo, *J. Am. Chem. Soc.*, 2015, **137**, 4347–4357.
- 3 K. Fuku, Y. Miyase, Y. Miseki, T. Gunji and K. Sayama, *ChemistrySelect*, 2016, **1**(18), 5721–5726.
- 4 J. A. Arminio-Ravelo, A. W. Jensen, K. D. Jensen, J. Quinson and M. Escudero-Escribano, *Chem. Phys. Chem.*, 2019, **20**(22), 2956–2963.
- 5 T. M. Gill, L. Vallez and X. Zheng, *ACS Energy Lett.*, 2021, **6**, 2854–2862.
- 6 X. Shi, S. Siahrostami, G.-L. Li, Y. Zhang, P. Chakthranont, F. Studt, T. F. Jaramillo, X. Zheng and J. K. Nørskov, *Nat. Commun.*, 2017, **8**(1), 701.
- 7 A. J. Bard, R. Parsons and J. Jordan, *Standard Potentials in Aqueous Solution*, Routledge, 1985.
- 8 M. Carmo, D. L. Fritz, J. Mergel and D. Stolten, *Int. J. Hydrogen Energy*, 2013, **38**(12), 4901–4934.
- 9 S. Shiva Kumar and V. Himabindu, *Mater. Sci. Energy Technol.*, 2019, **2**(3), 442–454.
- 10 C. C. L. McCrory, S. Jung, I. M. Ferrer, S. M. Chatman, J. C. Peters and T. F. Jaramillo, *J. Am. Chem. Soc.*, 2015, **137**(13), 4347–4357.
- 11 Z. Lei, T. Wang, B. Zhao, W. Cai, Y. Liu, S. Jiao, Q. Li, R. Cao and M. Liu, *Adv. Energy Mater.*, 2020, **10**, 1–18.
- 12 J. Tang, T. Zhao, D. Solanki, X. Miao, W. Zhou and S. Hu, *Joule*, 2021, **5**, 1432–1461.
- 13 S. Siahrostami, S. J. Villegas, A. H. B. Mostaghimi, S. Back, A. B. Farimani, H. Wang, K. A. Persson and J. Montoya, *ACS Catal.*, 2020, **10**, 7495–7511.
- 14 J. M. Campos-Martin, G. Blanco-Brieva and J. L. G. Fierro, *Angew. Chem., Int. Ed.*, 2006, **45**(42), 6962–6984.
- 15 T. Nakajima, A. Hagino, T. Nakamura, T. Tsuchiya and K. Sayama, *J. Mater. Chem. A*, 2016, **4**, 17809–17818.
- 16 Q. Mi, A. Zhanaidarova, B. S. Brunschwig, H. B. Gray and N. S. Lewis, *Energy Environ. Sci.*, 2012, **5**, 5694–5700.
- 17 K. Fuku, N. Wang, Y. Miseki, T. Funaki and K. Sayama, *ChemSusChem*, 2015, **8**, 1593–1600.
- 18 S. H. Ahn, J. Zhao, J. H. Kim and X. Zheng, *Electrochim. Acta*, 2017, **244**, 184–191.
- 19 A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder and K. A. Persson, *APL Mater.*, 2013, DOI: [10.1063/1.4812323](https://doi.org/10.1063/1.4812323).
- 20 K. A. Persson, B. Walldwick, P. Lazic and G. Ceder, *Phys. Rev. B*, 2012, **85**(23), 235438.
- 21 S. Cherevko, S. Geiger, O. Kasian, N. Kulyk, J.-P. Grote, A. Savan, B. R. Shrestha, S. Merzlikin, B. Breitbach, A. Ludwig and K. J. J. Mayrhofer, *Catal. Today*, 2016, DOI: [10.1016/j.cattod.2015.08.014](https://doi.org/10.1016/j.cattod.2015.08.014).

Author contributions

X. L. Z. conceptualized the study. S. J. V. performed the theoretical analysis (DFT calculations) under S. S.'s supervision. A. T. G. performed XAS and the FT-EXAFS fittings under D. S.'s supervision. Y. J. performed SEM and EDS. S. P. performed XRD. L. V. designed and performed all other experiments and assembled the figures and manuscript under X. L. Z.'s supervision. Q. W. helped with some of the electrochemical experiments. L. V., S. J. V., and A. T. G. were involved in writing the original draft. X. L. Z., L. V., S. J. V., S. S., A. T. G., and T. M. G. were involved in reviewing and editing.

Conflicts of interest

There are no conflicts of interest to declare.

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- 22 N.-T. Suen, S.-F. Hung, Q. Quan, N. Zhang, Y.-J. Xu and H. M. Chen, *Chem. Soc. Rev.*, 2017, **46**(2), 337–365.
- 23 J. H. Baek, T. M. Gill, H. Abroshan, S. Park, X. Shi, J. Nørskov, H. S. Jung, S. Siahrostami and X. Zheng, *ACS Energy Lett.*, 2019, **4**(3), 720–728.
- 24 X. Chen and S. S. Mao, *Chem. Rev.*, 2007, **107**(7), 2891–2959.
- 25 S.-Y. Lee and S.-J. Park, *J. Ind. Eng. Chem.*, 2019, **19**(6), 1761–1769.
- 26 M. García-Mota, A. Vojvodic, H. Metiu, I. C. Man, H.-Y. Su, J. Rossmeisl and J. K. Nørskov, *ChemCatChem*, 2011, **3**(10), 1607–1611.
- 27 A. Pedersen, J. Barrio, A. Li, R. Jarvis, D. J. L. Brett, M. M. Titirici and I. E. L. Stephens, *Adv. Energy Mater.*, 2021, **12**(3), 2102715.
- 28 H. Sudrajat, S. Babel, A. T. Ta and T. K. Nguyen, *J. Phys. Chem. Solids*, 2020, **144**, 109517.
- 29 L. Wang, X. Zhang, P. Zhang, Z. Cao and J. Hu, *J. Saudi Chem. Soc.*, 2015, **19**, 595–601.
- 30 Q. R. Deng, X. H. Xia, M. L. Guo, Y. Gao and G. Shao, *Mater. Lett.*, 2011, **65**, 2051–2054.
- 31 V. D. Binas, K. Sambani, T. Maggos, A. Katsanaki and G. Kiriakidis, *Appl. Catal., B*, 2012, **113–114**, 79–86.
- 32 R. Chauhan, A. Kumar and R. P. Chaudhary, *Spectrochim. Acta, Part A*, 2012, **98**, 256–264.
- 33 L. Wang, J. Fan, Z. Cao, Y. Zheng, Z. Yao, G. Shao and J. Hu, *Chem. – Asian J.*, 2014, **9**, 1904–1912.
- 34 X. H. Xia, L. Lu, A. S. Walton, M. Ward, X. P. Han, R. Brydson, J. K. Luo, G. Shao, X. H. Xia, L. Lu, A. S. Walton, M. Ward, X. P. Han, R. Brydson, J. K. Luo and G. Shao, *Acta Mater.*, 2012, **60**, 1974–1985.
- 35 H. Chen, T. Wu, X. Li, S. Lu, F. Zhang, Y. Wang, H. Zhao, Q. Liu, Y. Luo, A. M. Asiri, Z.-S. Feng, Y. Zhang and X. Sun, DOI: [10.1021/acssuschemeng.0c09009](https://doi.org/10.1021/acssuschemeng.0c09009).
- 36 Q. Chen, C. Ma, S. Yan, J. Liang, K. Dong, Y. Luo, Q. Liu, T. Li, Y. Wang, L. Yue, B. Zheng, Y. Liu, S. Gao, Z. Jiang, W. Li and X. Sun, *ACS Appl. Mater. Interfaces*, 2021, **13**, 46659–46664.
- 37 J. Li, D. Solanki, Q. Zhu, X. Shen, G. Callander, J. Kim, Y. Li, H. Wang and S. Hu, *J. Mater. Chem. A*, 2021, **9**, 18498–18505.
- 38 R. Katal, S. Masudy-Panah, M. Tanhaei, M. H. D. A. Farahani and H. Jiangyong, *Chem. Eng. J.*, 2020, **384**, 123384.
- 39 W. J. Ong, L. L. Tan, S. P. Chai, S. T. Yong and A. R. Mohamed, *Nanoscale*, 2014, **6**, 1946–2008.
- 40 I. C. Man, H. Y. Su, F. Calle-Vallejo, H. A. Hansen, J. I. Martínez, N. G. Inoglu, J. Kitchin, T. F. Jaramillo, J. K. Nørskov and J. Rossmeisl, *ChemCatChem*, 2011, **3**, 1159–1165.
- 41 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jónsson, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.
- 42 M. J. Craig, G. Coulter, E. Dolan, J. Soriano-López, E. Mates-Torres, W. Schmitt and M. García-Melchor, *Nat. Commun.*, 2019, **10**, 1–9.
- 43 L. C. Seitz, C. F. Dickens, K. Nishio, Y. Hikita, J. Montoya, A. Doyle, C. Kirk, A. Vojvodic, H. Y. Hwang, J. K. Nørskov and T. F. Jaramillo, *Science*, 2016, **353**, 1011LP–1014.
- 44 G. Gao, E. R. Waclawik and A. Du, *J. Catal.*, 2017, **352**, 579–585.
- 45 D. Banerjee, S. K. Gupta, N. Patra, S. Wasim Raja, N. Pathak, D. Bhattacharyya, P. K. Pujari, S. V. Thakare and S. N. Jha, *Phys. Chem. Chem. Phys.*, 2018, **20**, 28699.
- 46 D. Gonbeau, C. Guimon, G. Pfister-Guillouzo, A. Levasseur, G. Meunier and R. Dormoy, *Surf. Sci.*, 1991, **254**, DOI: [10.1016/0039-6028\(91\)90640-E](https://doi.org/10.1016/0039-6028(91)90640-E).
- 47 C. Sleight, A. P. Pijpers, A. Jaspers, B. Coussens and R. J. Meier, *J. Electron Spectrosc. Relat. Phenom.*, 1996, **77**(1), 41–57.
- 48 M. Guo, Y. Gao and G. Shao, *Phys. Chem. Chem. Phys.*, 2016, **18**(4), 2818–2829.
- 49 I. Saratovsky, P. G. Wightman, P. A. Pastén, J.-F. Gaillard and K. R. Poeppelmeier, *J. Am. Chem. Soc.*, 2006, **128**, 11188–11198.
- 50 E. Chalmin, F. Farges and G. E. Brown, *Contrib. Mineral. Petrol.*, 2008, **157**, 111–126.
- 51 D. Joseph, A. K. Yadav, S. N. Jha and D. Bhattacharyya, *Bull. Mater. Sci.*, 2013, **36**(6), DOI: [10.1007/s12034-013-0567-8](https://doi.org/10.1007/s12034-013-0567-8).
- 52 V. Celorrio, L. Calvillo, G. Granozzi, A. E. Russell and D. J. Fermin, *Top. Catal.*, 2018, **61**, 154–161.
- 53 P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. Dal Corso, S. De Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, A. Smogunov, P. Umari and R. M. Wentzcovitch, *J. Phys.: Condens. Matter*, 2009, **21**(39), 395502.
- 54 S. R. Bahn and K. W. Jacobsen, *Comput. Sci. Eng.*, 2002, **4**, 56–66.
- 55 T. Wen, J. Gao, J. Shen and Z. Zhou, *J. Mater. Sci.*, 2001, **36**(24), 5923–5926.
- 56 B. Ravel and M. Newville, *J. Synchrotron Radiat.*, 2005, **12**, 537–541.
- 57 W. Xing, M. Yin, Q. Lv, Y. Hu, C. Liu and J. Zhang, *Rotating Electrode Methods Oxyg. Reduct. Electrocatal.*, 2014, 1–31.
- 58 M. H. Mariano, *Anal. Chim. Acta*, 1962, **17**, 16.
- 59 D. Belhatche and J. M. Symons, *J. – Am. Water Works Assoc.*, 1991, **83**(8), 70–73.
- 60 T. M. Gill and X. Zheng, *Chem. Mater.*, 2020, **32**(15), 6285–6294.

