

# Optimally tuned starting point for single-shot $GW$ calculations of solids

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The dependence of *ab initio* many-body perturbation theory within the  $GW$  approximation on the eigensystem used in calculating quasiparticle corrections limits this method's predictive power. Here, we investigate the accuracy of the recently developed Wannier-localized optimally tuned screened range-separated hybrid (WOT-SRSH) functional as a generalized Kohn-Sham starting point for single-shot  $GW$  ( $G_0W_0$ ) calculations for a range of semiconductors and insulators. Comparison to calculations based on well-established functionals, namely, PBE, PBE0, and HSE, as well as to self-consistent  $GW$  schemes and to experiment, shows that band gaps computed via  $G_0W_0$ @WOT-SRSH have a level of precision and accuracy that is comparable to that of more advanced methods such as quasiparticle self-consistent  $GW$  and eigenvalue self-consistent  $GW$ . We also find that  $G_0W_0$ @WOT-SRSH improves the description of states deeper in the valence band manifold. Finally, we show that  $G_0W_0$ @WOT-SRSH significantly reduces the sensitivity of computed band gaps to ambiguities in the underlying WOT-SRSH tuning procedure.

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## I. INTRODUCTION

*Ab initio* many-body perturbation theory within the  $GW$  approximation is a state-of-the-art approach for calculations of the quasiparticle (QP) band structures of crystalline solids [1–16]. In the  $GW$  approximation, the self-energy  $\Sigma$  is given by the convolution  $\Sigma = iGW$ , where  $G$  is the single particle Green's function and  $W$  is the dynamically screened Coulomb interaction. The  $GW$  self-energy is normally first constructed from a (generalized) Kohn-Sham GKS [17] “starting point,” an eigensystem computed from density functional theory (DFT). While semilocal functionals, such as the local density approximation [18] or generalized gradient approximations like PBE [19], have historically been the standard choice for constructing this starting point eigensystem [5,8,9,11], hybrid functionals are increasingly used [16,20–28].

In practice, there are a variety of choices regarding how  $GW$  calculations are carried out, with significant consequences for accuracy [16,29]. Once the  $GW$  self-energy has been constructed, the quasiparticle energies can be computed via first-order correction to the GKS eigenvalues, the so-called single-shot  $GW$  ( $G_0W_0$ ) approach [5], or  $G$  and/or  $W$  can be iterated to self-consistency [21,22,24–26,28,30–50]. The  $G_0W_0$  method is the computationally least expensive approach, and, as has been well established, the quasiparticle band structures computed with  $G_0W_0$  approaches typically substantially improve agreement with experiment compared

to those obtained directly from the GKS eigenvalues of their underlying DFT starting points [16,20,44,49,51]. For example, QP band gap data from an analysis of  $G_0W_0$  calculations for various semiconductors and insulators by Grumet *et al.* [49] exhibited a mean absolute error (MAE) of 0.2 eV compared to an MAE of 1.2 eV for the underlying DFT functionals used. However,  $G_0W_0$  results exhibit a starting point dependence, where results can depend considerably on the DFT functional used to construct the starting eigensystem [12,16,20,27,52]. For molecules, a range of about 1 eV in  $G_0W_0$  calculations of highest occupied molecular orbital energies has been reported [53,54]. Likewise,  $G_0W_0$  results for solids (e.g., Si, InN, ZnO, ZnS, CdS, and GaN) [20,55] have shown a similar starting point dependence of up to 2 eV in computed band gaps. As such, it is common practice to differentiate  $G_0W_0$  calculations by the functional used in their starting point, denoted by  $G_0W_0$ @(...). Relatedly, the accuracy of  $G_0W_0$  calculations based on semilocal DFT functionals is known to depend on a fortuitous and sometimes unreliable cancellation of error between the lack of consideration of vertex corrections, which tends to cause underscreening in  $W_0$ , and the systematic underestimation of band gaps computed from semilocal functionals, which tends to cause overscreening [35,36].

One way to address the issue of starting point dependence is to construct the self-energy in a more self-consistent manner, leading to the development of methods like eigenvalue self-consistent  $GW$  (evGW) [6,32,35] and quasiparticle self-consistent  $GW$  (QSGW) [33,34,36]. In evGW, the eigenvalues used to construct  $G$  and  $W$  are iterated to

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self-consistency. Though *evGW* is noticeably less dependent on the starting point used, the wave functions used in constructing  $G$  and  $W$  are not updated in this approach, leading to a residual modest starting point dependence (e.g., 0.4 eV in the case of azabenzenes [53]). On the other hand, *QSGW* seeks to variationally minimize the difference between the self-energy and a static nonlocal potential by updating both the wave functions and eigenvalues used to construct  $G$  and  $W$  and has been shown to be mostly independent of the starting point used [56] (though there do exist questions as to whether this holds true for some metal oxides [57,58]). While iterating on  $G$  and/or  $W$  provides more consistent results, it also requires greater computational resources. Additionally, while the self-consistent correction of the QP eigenvalues accounts for the error due to DFT band gap underestimation in these methods, it does not systematically account for the lack of vertex corrections, leading to underscreening and larger QP band gaps [35,37]. For example, Grunet *et al.* report that *evGW* and *QSGW* overestimate QP gaps by 1.0 and 0.8 eV on average, respectively [49].

While *GW* self-consistency schemes can reduce the starting point dependence of  $G_0W_0$ , the increased cost of going beyond  $G_0W_0$  has incentivized the development of starting points for  $G_0W_0$  calculations which do not suffer from the same level of starting point dependence [20–22,24,25,27,28,55]. In particular, hybrid DFT functionals, which include exact exchange, are an appealing candidate for improved  $G_0W_0$  starting points for multiple reasons. For example, the GKS band gaps computed with these functionals vary with the amount of exact exchange present, and therefore can be used to remedy the overscreening due to band gap underestimation that is present in semilocal functionals [16]. Moreover, hybrid functionals can better address the starting point dependence associated with more localized  $d$  states [35,55,59], where self-interaction errors present in semilocal functionals are more pronounced [60] and lead to spurious orbital energy ordering that can propagate to the *GW* eigenspectrum [16]. In such cases, the presence of exact exchange can help to reduce this error [21,28,53,55,61,62].

The use of hybrid functionals like PBE0 [63] and HSE [64] as starting points for  $G_0W_0$  calculations has been shown to generally improve agreement with experiment [20,27]. Moreover, some hybrid functionals can be tuned [65] to satisfy the ionization potential (IP) theorem [66,67], suggesting the possibility of a more physically accurate and consistent starting point eigensystem. Specifically, Wing *et al.* [68] developed a procedure for parametrizing a class of screened range-separated hybrid (SRSH) functionals capable of accurately predicting the band gaps of solid-state materials without empirical parameters, directly from density functional theory. The parametrization is arrived at by capturing the asymptotic limit of the screened exchange potential and by using an ansatz based on the IP theorem which applies to localized Wannier functions in systems with periodic boundary conditions [69]. This class of Wannier-localized optimally tuned screened range-separated hybrid (WOT-SRSH) functionals has been recently used to calculate the fundamental band gaps of semiconductors and insulators, leading to excellent agreement with experiment, with an MAE of 0.1 eV [68].

TABLE I. Hybrid functionals in the SRSH formalism.

|          | $\alpha$            | $\alpha + \beta$          | $\gamma$ ( $a_0^{-1}$ ) |
|----------|---------------------|---------------------------|-------------------------|
| PBE0     | 0.25                | 0.25                      | 0                       |
| HSE06    | 0.25                | 0                         | 0.106                   |
| WOT-SRSH | Varies <sup>a</sup> | $\varepsilon_\infty^{-1}$ | Tuned                   |

<sup>a</sup>By default  $\alpha$  is set to be 0.25, but in cases where  $\alpha + \beta \approx 0.25$  the value of  $\alpha$  is increased slightly until the IP ansatz can be satisfied. For more details see [68] or the discussions in Secs. II A and IV B.

For molecules, the use of optimally tuned range-separated hybrid functionals which enforce the IP theorem as a starting point for  $G_0W_0$ , as suggested in [23], has been shown to be successful [25,42,43,47,70,71]. However, as of yet, there has not been an analogous exploration of these nonempirical WOT-SRSH starting points which approximately satisfy the IP theorem for  $G_0W_0$  calculations of solid-state systems.

Here, we undertake such an exploration and analyze the performance of single-shot  $G_0W_0$ @WOT-SRSH calculations. For a series of 15 semiconductor and insulators, we construct  $G_0$  and  $W_0$  using WOT-SRSH and compute band gaps as well as properties associated with states deeper in the valence band manifold such as valence bandwidths and  $d$  band positions. We then compare results with experiments and calculations from other DFT starting points. We also discuss how  $G_0W_0$  corrections affect the sensitivity of computed bands gaps to ambiguities in the WOT-SRSH tuning procedure. Overall, our calculations demonstrate that a  $G_0W_0$ @WOT-SRSH approach provides accurate quasiparticle properties for a broad range of materials, opening the door to predictive single-shot  $G_0W_0$  calculations for chemically complex solids.

## II. THEORY

### A. DFT

The starting point for our *GW* calculations are GKS orbitals  $\phi_{nk}$  and eigenenergies  $\epsilon_{nk}^0$ , where  $n$  is the band index and  $k$  the wave vector. Here, we primarily focus on the SRSH functional scheme [39,72–74]. This class of functionals is formulated by partitioning the exchange portion of the Coulomb potential into

$$\frac{1}{r} = \frac{\alpha + \beta \text{erf}(\gamma r)}{r} + \frac{1 - [\alpha + \beta \text{erf}(\gamma r)]}{r}. \quad (1)$$

This partition introduces three parameters  $\alpha$ ,  $\beta$ , and  $\gamma$ , the physical and computational significance of which is discussed shortly. When implemented in the hybrid functional, the first term of Eq. (1) is treated explicitly with Fock exchange, whereas the second term is replaced with an approximate semilocal exchange functional [39,72]. In this framework,  $\alpha$  regulates the amount of exact exchange in the short range,  $\alpha + \beta$  regulates the amount of exact exchange in the long range, and  $\gamma$  is the length scale for the transition between these two limits. The correlation component is treated with the same functional used for the semilocal part of the aforementioned exchange partition. By specifying the values of  $\alpha$ ,  $\alpha + \beta$ , and  $\gamma$ , we can recover various well-known hybrid functionals (Table I). For example, if the semilocal exchange is based on the PBE functional [19], then setting  $\gamma = 0$  produces a global

TABLE II. Parameters used in the DFT starting point calculations. Lattice parameters were taken from experiment, and WOT-SRSH parameters were taken from prior work [68].

|      | Lattice parameters |                   |                     | WOT-SRSH parameters <sup>d</sup> |         |                         |
|------|--------------------|-------------------|---------------------|----------------------------------|---------|-------------------------|
|      | $a$                | $c$               | $u$                 | $\alpha$                         | $\beta$ | $\gamma$ ( $a_0^{-1}$ ) |
| InSb | 6.48 <sup>a</sup>  |                   |                     | 0.25                             | -0.1745 | 0.17                    |
| InAs | 6.06 <sup>a</sup>  |                   |                     | 0.25                             | -0.1623 | 0.16                    |
| Ge   | 5.66 <sup>a</sup>  |                   |                     | 0.25                             | -0.1824 | 0.19                    |
| GaSb | 6.1 <sup>a</sup>   |                   |                     | 0.25                             | -0.1733 | 0.19                    |
| Si   | 5.43 <sup>a</sup>  |                   |                     | 0.25                             | -0.1611 | 0.24                    |
| InP  | 5.87 <sup>a</sup>  |                   |                     | 0.25                             | -0.1373 | 0.23                    |
| GaAs | 5.65 <sup>a</sup>  |                   |                     | 0.25                             | -0.1549 | 0.15                    |
| AlSb | 6.14 <sup>a</sup>  |                   |                     | 0.25                             | -0.1482 | 0.14                    |
| AlAs | 5.66 <sup>a</sup>  |                   |                     | 0.3                              | -0.1779 | 0.18                    |
| GaP  | 5.45 <sup>a</sup>  |                   |                     | 0.25                             | -0.1375 | 0.21                    |
| AlP  | 5.47 <sup>a</sup>  |                   |                     | 0.25                             | -0.1128 | 0.16                    |
| C    | 3.57 <sup>a</sup>  |                   |                     | 0.3                              | -0.1198 | 0.23                    |
| AlN  | 3.11 <sup>a</sup>  | 4.98 <sup>a</sup> | 0.3821 <sup>c</sup> | 0.35                             | -0.1073 | 0.26                    |
| MgO  | 4.22 <sup>a</sup>  |                   |                     | 0.25                             | 0.0948  | 1.5                     |
| LiF  | 4.03 <sup>b</sup>  |                   |                     | 0.25                             | 0.2681  | 1.08                    |

<sup>a</sup>[81].

<sup>b</sup>[82].

<sup>c</sup>[83].

<sup>d</sup>[68].

hybrid functional, PBE $\alpha$  [75,76], and if  $\alpha = 0.25$ , PBE0 is obtained. For  $\gamma = 0.106 a_0^{-1}$ , setting  $\alpha + \beta = 0$  and  $\alpha = 0.25$  yields the HSE functional [64].

In this paper, we focus on the novel WOT-SRSH formulation [68] of the SRSH functional. Here, the choice  $\alpha + \beta = \varepsilon_{\infty}^{-1}$ , where  $\varepsilon_{\infty}^{-1}$  is the orientationally averaged electronic contribution to the dielectric constant, enforces the asymptotically correct long-range screening in the Coulomb potential [39,73,74]. The range-separation parameter  $\gamma$  is nonempirically selected by enforcing an ansatz which extends the IP theorem to the removal of an electron from the highest-energy occupied maximally localized Wannier function (MLWF) [69]. By default, we choose  $\alpha = 0.25$  because, as seen in global hybrids, setting  $\alpha = 0.25$  has proven effective for many molecular and solid-state systems [41,75,77,78]. In cases where setting  $\alpha = 0.25$  does not yield a unique choice of  $\gamma$  via the IP ansatz, as is often the case when  $\varepsilon_{\infty}^{-1} \approx 0.25$ ,  $\alpha$  is increased slightly until an optimal value of  $\gamma$  that does not approach zero can be found. The need for a lower bound on the size of  $\gamma$  is related to the “ $\gamma$  collapse problem” [79,80], where small values of  $\gamma$  result in an unphysical effectively PBE $\alpha$  hybrid functional if  $\gamma^{-1}$  exceeds the size of the unit cell of the calculation. As seen in Table II, the largest value that  $\alpha$  needed to be increased to was 0.35. With these constraints, WOT-SRSH functionals are a system-specific but nonempirical class of exchange correlation (xc) functionals that result in a GKS eigensystem that consistently and accurately predicts the QP band gaps of solids, compensating by construction for the derivative discontinuity error present in most density functionals [11,17,84–89].

### B. GW method

In the *ab initio* GW approach, the self-energy  $\Sigma = iGW$  of a system is constructed from a DFT GKS eigensystem. As

discussed, this GKS eigensystem  $\{\phi_{nk}, \epsilon_{nk}^{\text{DFT}}\}$  depends on the underlying xc functional  $V_{xc}$  used to compute it, and by extension the self-energy computed from this eigensystem is also sensitive to the choice of  $V_{xc}$ . Specifically, the single-particle Green’s function  $G_0$  is constructed as

$$G_0(\mathbf{r}, \mathbf{r}'; \omega) = \sum_{nk} \frac{\phi_{nk}(\mathbf{r})\phi_{nk}^*(\mathbf{r}')}{\omega - \epsilon_{nk}^{\text{DFT}} \pm i\eta}, \quad (2)$$

where  $\eta$  is a positive infinitesimal real number, and the  $\pm$  in front of it is negative for occupied states and positive for empty states. The dynamically screened Coulomb interaction  $W_0$  is given by

$$W_0(\mathbf{r}, \mathbf{r}'; \omega) = \int d\mathbf{r}'' \varepsilon^{-1}(\mathbf{r}, \mathbf{r}''; \omega) v(\mathbf{r}', \mathbf{r}''), \quad (3)$$

where  $v(\mathbf{r}, \mathbf{r}') = |\mathbf{r} - \mathbf{r}'|^{-1}$  and where the dielectric function,

$$\varepsilon^{-1}(\mathbf{r}, \mathbf{r}'; \omega) = \delta(\mathbf{r}, \mathbf{r}') - \int d\mathbf{r}'' v(\mathbf{r}, \mathbf{r}'') \chi_0(\mathbf{r}'', \mathbf{r}', \omega), \quad (4)$$

is computed within the random-phase approximation (RPA) [6] based on the polarizability  $\chi_0(\mathbf{r}, \mathbf{r}', \omega)$ , given by the Adler-Wiser expression [90,91]

$$\chi_0(\mathbf{r}, \mathbf{r}', \omega) = \sum_{nk} \sum_{n'k'}^{\text{occ. emp.}} \left[ \frac{\phi_{nk}^*(\mathbf{r})\phi_{n'k'}(\mathbf{r})\phi_{nk}^*(\mathbf{r}')\phi_{n'k'}(\mathbf{r}')}{\omega - (\epsilon_{n'k'}^{\text{DFT}} - \epsilon_{nk}^{\text{DFT}}) + i\eta} - \frac{\phi_{nk}(\mathbf{r})\phi_{n'k'}^*(\mathbf{r})\phi_{nk}(\mathbf{r}')\phi_{n'k'}^*(\mathbf{r}')}{\omega + (\epsilon_{n'k'}^{\text{DFT}} - \epsilon_{nk}^{\text{DFT}}) - i\eta} \right], \quad (5)$$

where the summations are over the occupied and unoccupied bands. In practice,  $\chi_0(\mathbf{r}, \mathbf{r}', \omega)$  is often evaluated statically ( $\omega = 0$ ), and a simplified model frequency dependence, such as a plasmon pole model (PPM), is used instead [6,92,93]. We also note that a consideration of the denominators in Eq. (5) clarifies why the under- or overestimation of the band gap can result in over- or underscreening in  $W_0$ , respectively.

With the above quantities, the  $G_0W_0$  self-energy becomes

$$\Sigma(\mathbf{r}, \mathbf{r}'; \omega) = \frac{i}{2\pi} \int d\omega' G_0(\mathbf{r}, \mathbf{r}'; \omega + \omega') W_0(\mathbf{r}, \mathbf{r}'; \omega') \times e^{i\omega'\eta}. \quad (6)$$

This  $G_0W_0$  operator can then be used to correct the DFT eigenvalues perturbatively via

$$\epsilon_{nk}^{\text{QP}} = \epsilon_{nk}^{\text{DFT}} + \langle nk | \Sigma(\epsilon_{nk}^{\text{QP}}) - V_{xc} | nk \rangle, \quad (7)$$

where, to avoid double counting of beyond-Hartree interactions, the contributions of  $V_{xc}$  are subtracted off. Due to the fact that  $\epsilon_{nk}^{\text{QP}}$  in Eq. (7) depends on itself, evaluating this expression is nontrivial. However, as is common practice [94–96], we expand Eq. (7) to first order about  $\epsilon_{nk}^{\text{DFT}}$  to evaluate it efficiently.

## III. COMPUTATIONAL DETAILS

### A. DFT calculations

Our DFT calculations are performed using a modified version of the QUANTUM ESPRESSO (version 6.2) plane-wave

code [97–99] that allows for the use of the SRSB functional [39] of Eq. (1) with arbitrary  $\alpha$ ,  $\beta$ , and  $\gamma$  parameters. Other modifications also allow for a more efficient calculation of many hundreds of unoccupied states for GKS systems using adaptively compressed exchange [100] via what amounts to a non-self-consistent field calculation once the occupied orbitals and ground state density have been converged (see Supplemental Material Sec. S-I [101] for more details). All calculations utilize fully relativistic optimized norm-conserving Vanderbilt pseudopotentials [102] obtained from the PSEUDO-DOJO repository [103]. Using these pseudopotentials, the effects of spin-orbit coupling (SOC) are included self-consistently at the DFT level for all calculated observables. For Ge, Ga, In, Sb, and As, the electrons within a complete set of semicore shells of the same principal quantum number are treated as valence electrons. For calculations using hybrid functionals and the GW methods, the explicit consideration of these deeper states has been shown to be necessary for the accurate description of the electronic structure of such systems [32,104–106]. A plane-wave energy cutoff of 135 Ry and experimental room temperature lattice parameters (summarized in Table II) are used for all systems.

For hybrid functionals, the energy cutoff involved in constructing the exact exchange operator is lowered, without significantly impacting the results at the DFT or  $G_0W_0$ @DFT levels, from its default value of four times the plane-wave energy cutoff to 150 Ry. In some rare cases where this causes numerical instability in the self-consistent evaluation of the exchange, namely, computing the PBE0 starting points for Ge and InAs, this cutoff is raised to the default value of four times the plane-wave energy cutoff.

### B. GW calculations

All our GW calculations are carried out using the BERKELEYGW package [6,107]. In an effort to minimize the cost of computing many hundreds of unoccupied states using hybrid functionals, the dielectric function is constructed using a symmetry-reduced unshifted Monkhorst-Pack  $\mathbf{q}$  grid. Frequency dependence in the dielectric function is included approximately via the Godby-Needs PPM [92,93], which has been shown to reproduce the computed band gaps of full-frequency integration at reduced cost [108]. It should be noted, however, that this comparable level of accuracy can wane for deeper valence states; previous studies [109,110] report that valence bandwidths and  $d$  band binding energies computed using the Godby-Needs PPM are modestly overestimated relative to full-frequency integration.

The static remainder approximation to  $\Sigma$  [111] is used whenever it yields faster convergence with respect to the number of bands, which is the case for all materials except AlN, MgO, and LiF. The band gaps of all materials are converged within (or well within) 50 meV with respect to the number of bands used to construct  $\epsilon$  and  $\Sigma$ , the energy cutoff in the construction of  $\epsilon$ , and the unshifted  $\mathbf{k}$  grid being used. For more convergence details, see Supplemental Material Sec. S-II [101].

The effects of SOC are computed at the DFT level and added perturbatively at the  $G_0W_0$  level for all materials, an approximation which has precedent and justification for the

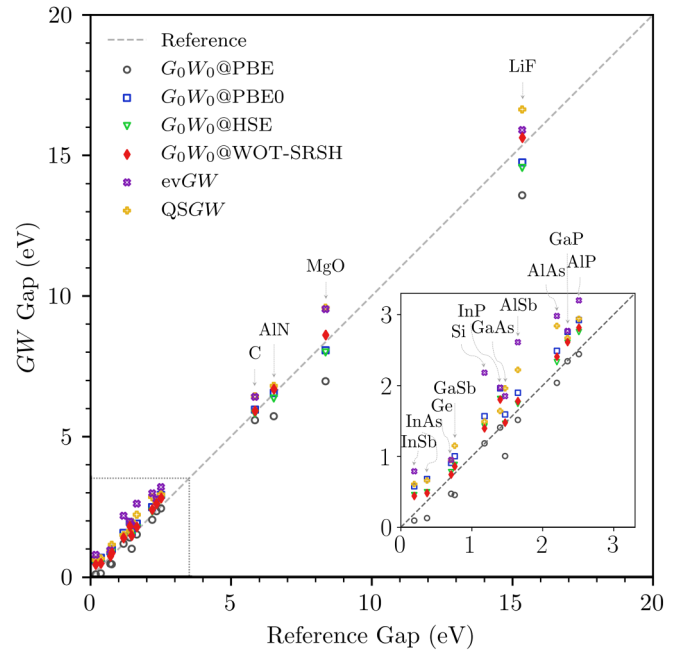


FIG. 1. QP band gaps computed using  $G_0W_0$ @PBE,  $G_0W_0$ @PBE0,  $G_0W_0$ @HSE,  $G_0W_0$ @WOT-SRSH,  $evGW$ , and QSGW in reference to ZPR corrected experimental results. Data are taken from Table III. The inset in the lower right corner is a zoom-in of the bottom 3 eV of the data set.

classes of materials under study [112–114]. While BERKELEYGW does allow for the explicit computation of SOC effects at the GW level, this would require twice as many bands in the starting point eigensystem, quadrupling the cost of already expensive calculations. However, we find the error of including SOC perturbatively to be minimal. For example we report that for AlSb, a system with a strong SOC band gap renormalization of 240 meV, the error in the computed SOC renormalization of the band gap is only 6 meV. For systems with weaker renormalizations like GaP, this error is only 1 meV.

For band structures with conduction band minima off high symmetry points (as is the case for Si, C, GaP, and AlSb), eigenvalues are interpolated using the WANNIER90 code [115]. Due to the similarity in orbital character of the states near the band gap for all the aforementioned systems, only the four highest occupied and four lowest unoccupied bands about the band edges are Wannierized, with  $sp^3$  starting projections being used for all of them. SOC corrections to the interpolated bands are determined for each eigenenergy  $\epsilon_{nk}$  and interpolated using MLWFs, as outlined by Malone and Cohen [112].

## IV. RESULTS AND DISCUSSION

### A. Quasiparticle band structures

#### 1. Band gaps

The QP band gaps of 15 semiconductors and insulators calculated from WOT-SRSH, as well as  $G_0W_0$  using four starting points (PBE, PBE0, HSE06, and WOT-SRSH), are given in Table III and graphed in Fig. 1. Additional results from DFT functionals other than WOT-SRSH can be found in



TABLE III. QP band gaps (in eV) at the WOT-SRSH and  $G_0W_0$ @DFT level for the various compounds and functionals under study. At the bottom of the table are the MAE (mean absolute error), MSE (mean signed error), and max error; all are in eV and measured relative to the reported reference values, which are arrived at by incorporating ZPR corrections into experimental band gap data. Experimental results are arrived at via an analysis of optical absorption spectroscopy data, where excitonic effects are taken into account to arrive at the fundamental gap (see [68] for details).

|           | $G_0W_0$ @ PBE | $G_0W_0$ @ PBE0 | $G_0W_0$ @ HSE | WOT-SRSH | $G_0W_0$ @ WOT-SRSH | evGW               | QSGW               | Ref   | Expt, ZPR                              |
|-----------|----------------|-----------------|----------------|----------|---------------------|--------------------|--------------------|-------|--|
| InSb      | 0.09           | 0.58            | 0.45           | 0.32     | 0.44                | 0.79 <sup>a</sup>  | 0.61 <sup>a</sup>  | 0.19  | 0.17 <sup>e</sup> , 0.02 <sup>j</sup>  |
| InAs      | 0.13           | 0.68            | 0.50           | 0.42     | 0.48                |                    | 0.66 <sup>c</sup>  | 0.37  | 0.35 <sup>e</sup> , 0.02 <sup>j</sup>  |
| Ge        | 0.47           | 0.91            | 0.78           | 0.69     | 0.74                | 0.95 <sup>b</sup>  | 0.95 <sup>c</sup>  | 0.71  | 0.66 <sup>f</sup> , 0.05 <sup>j</sup>  |
| GaSb      | 0.46           | 1.00            | 0.88           | 0.69     | 0.86                |                    | 1.15 <sup>c</sup>  | 0.76  | 0.73 <sup>e</sup> , 0.03 <sup>j</sup>  |
| Si        | 1.18           | 1.57            | 1.42           | 1.14     | 1.40                | 2.18 <sup>a</sup>  | 1.49 <sup>a</sup>  | 1.18  | 1.12 <sup>f</sup> , 0.06 <sup>j</sup>  |
| InP       | 1.41           | 1.96            | 1.81           | 1.56     | 1.80                | 1.97 <sup>a</sup>  | 1.64 <sup>a</sup>  | 1.40  | 1.35 <sup>e</sup> , 0.05 <sup>j</sup>  |
| GaAs      | 1.01           | 1.59            | 1.46           | 1.41     | 1.48                | 1.85 <sup>b</sup>  | 1.96 <sup>c</sup>  | 1.47  | 1.42 <sup>e</sup> , 0.05 <sup>j</sup>  |
| AlSb      | 1.51           | 1.90            | 1.74           | 1.71     | 1.78                | 2.61 <sup>a</sup>  | 2.22 <sup>a</sup>  | 1.65  | 1.61 <sup>e</sup> , 0.04 <sup>j</sup>  |
| AlAs      | 2.04           | 2.49            | 2.33           | 2.25     | 2.41                | 2.98 <sup>a</sup>  | 2.84 <sup>a</sup>  | 2.20  | 2.16 <sup>e</sup> , 0.04 <sup>j</sup>  |
| GaP       | 2.34           | 2.75            | 2.60           | 2.39     | 2.61                | 2.77 <sup>a</sup>  | 2.67 <sup>a</sup>  | 2.35  | 2.27 <sup>e</sup> , 0.08 <sup>j</sup>  |
| AlP       | 2.44           | 2.92            | 2.75           | 2.52     | 2.82                | 3.2 <sup>a</sup>   | 2.94 <sup>a</sup>  | 2.51  | 2.49 <sup>e</sup> , 0.02 <sup>j</sup>  |
| C         | 5.58           | 5.95            | 5.82           | 5.76     | 5.92                | 6.41 <sup>a</sup>  | 6.43 <sup>a</sup>  | 5.85  | 5.47 <sup>g</sup> , 0.38 <sup>k</sup>  |
| AlN       | 5.72           | 6.55            | 6.35           | 6.56     | 6.69                |                    | 6.80 <sup>c</sup>  | 6.52  | 6.14 <sup>e</sup> , 0.38 <sup>k</sup>  |
| MgO       | 6.96           | 8.07            | 7.99           | 8.16     | 8.62                | 9.53 <sup>a</sup>  | 9.58 <sup>a</sup>  | 8.36  | 7.83 <sup>h</sup> , 0.53 <sup>l</sup>  |
| LiF       | 13.58          | 14.75           | 14.55          | 15.34    | 15.63               | 15.90 <sup>b</sup> | 16.63 <sup>d</sup> | 15.35 | 14.20 <sup>i</sup> , 1.15 <sup>l</sup> |
| MAE       | 0.40           | 0.31            | 0.22           | 0.07     | 0.19                | 0.66 <sup>m</sup>  | 0.51               |       |  |
| MSE       | -0.40          | 0.19            | 0.038          | 0.003    | 0.19                | 0.66 <sup>m</sup>  | 0.51               |       |  |
| Max error | -1.77          | -0.60           | -0.80          | -0.20    | 0.40                | 1.17 <sup>m</sup>  | 1.28               |       |  |

<sup>a</sup>[49]; <sup>b</sup>[37]; <sup>c</sup>[34]; <sup>d</sup>[48]; <sup>e</sup>[116]; <sup>f</sup>[81]; <sup>g</sup>[117]; <sup>h</sup>[118]; <sup>i</sup>[119]; <sup>j</sup>[120]; <sup>k</sup>[121]; <sup>l</sup>[122,123]; <sup>m</sup>The MAE, MSE, and max error for evGW were computed using the available data for 12 out of 15 compounds.

Supplemental Material Sec. S-III.A [101]. While zero point renormalization (ZPR) effects due to electron-phonon coupling have a significant impact on the band gaps of many solids [124–128], they are not addressed computationally in this paper. Instead, computed band gaps are compared to reference band gaps which remove ZPR effects from the experimental measurements (see [68]). Additionally, excitonic effects are accounted for in our reference set by adding estimated or calculated exciton binding energies to the measured optical absorption edge or by inferring the fundamental band gap position based on the location and identification of excitonic absorption peaks in experimental data (see [68] for more details).

In line with what we have reported previously [68], the WOT-SRSH functional yields an excellent MAE of 0.07 eV and a mean signed error (MSE) of 0.00 eV for band gaps—the highest accuracy of all of the methods under study for this set of solids. As the MSE indicates, the data are nearly equally spread between over- and underestimating band gaps. Also, unlike the other functionals, WOT-SRSH has accuracy that is maintained for wider-band-gap systems and has a much smaller maximum magnitude error of 0.2 eV.

Performing  $G_0W_0$  based on the WOT-SRSH starting point for this set of materials yields an MAE of 0.19 eV, with the  $G_0W_0$ @WOT-SRSH calculated band gaps maintaining a similar level of precision with a maximum error of 0.40 eV. Notably, the  $G_0W_0$ @WOT-SRSH band gaps are all slightly overestimated, consistent with the overestimation observed with more rigorously self-consistent methods such as evGW and QSGW [34,37,48,49] (see Table III and Fig. 1). Some of the reported overestimation for these methods has been attributed to the absence of ZPR effects in the band gap

[34,37], but our reference band gap accounts for ZPR effects and still indicates some systematic overestimation. However, it is also known that the RPA dielectric function can underscreen and thus overestimate band gaps. As previously noted [37,45,48,129–131], beyond-RPA vertex corrections for a similar set of semiconductors and insulators can provide an improvement in the accuracy of the screening and QP band gaps once a consistent starting point that no longer underestimates the band gap is reached.

Comparing  $G_0W_0$ @WOT-SRSH to self-consistent GW approaches in Fig. 1, we find excellent agreement and superior performance relative to experiment for the systems studied here, at a lower computational cost.  $G_0W_0$ @WOT-SRSH also has a similar qualitative performance to these methods, consistently modestly overestimating band gaps across a broad range of materials.

As is well known and in agreement with prior work [16,20,44,49,51],  $G_0W_0$ @PBE significantly improves the accuracy of PBE band gaps, in this case bringing its MAE from 1.5 to 0.4 eV.  $G_0W_0$ @PBE also corrects major qualitative issues such as the inverted band gaps of InSb, InAs, and GaSb. Notably, however, band gaps of some insulators are still underestimated by more than 1 eV (e.g., MgO, LiF) by  $G_0W_0$ @PBE, leading to a substantial max error of -1.77 eV.

For the insulators studied (C, AlN, MgO, and LiF) we find that well-established hybrid functionals such as HSE and PBE0 offer a significant improvement over PBE as starting points for  $G_0W_0$  calculations. However, these hybrids perform slightly worse than PBE for the remaining semiconductors. Overall, for  $G_0W_0$ @PBE0, we observe a MAE of 0.31 eV, an MSE of 0.19 eV, and a max error of -0.60 eV. The smaller MSE relative to the MAE reflects that the band gaps

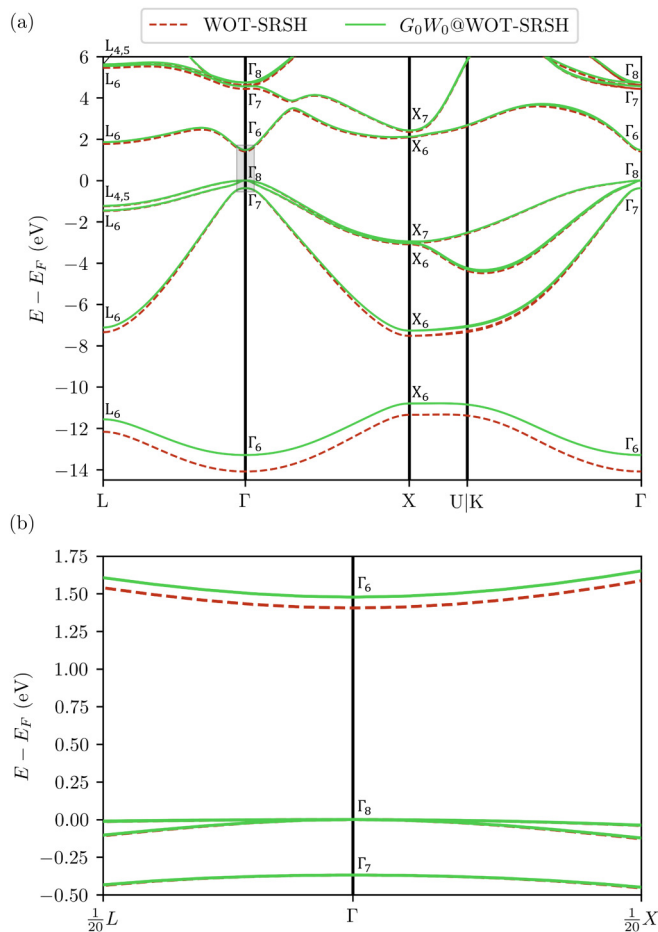


FIG. 2. Band structure of GaAs, including SOC, from WOT-SRSH and  $G_0W_0$ @WOT-SRSH. (a) A full plot of the band structure, including the eight highest occupied orbitals and the first few eV of the next eight unoccupied orbitals. (b) A zoomed-in inset—depicted by a gray box in (a)—of the valence and conduction band extrema. For both plots,  $E_F$  is taken to be the energy of the valence band maxima

computed from  $G_0W_0$ @PBE0 switch from being overestimated for small-band-gap semiconductors to being underestimated for wide-band-gap insulators.

In the case of  $G_0W_0$ @HSE, calculated QP band gaps have an MAE of 0.22 eV and an MSE of 0.04 eV. This comparatively near-zero MSE reflects that the band gaps computed from  $G_0W_0$ @HSE switch from being overestimated for small-band-gap semiconductors to being underestimated for wide-band-gap insulators. Moreover, while the MAE calculated for  $G_0W_0$ @HSE is quite low and comparable to what is seen for  $G_0W_0$ @WOT-SRSH, the performance of  $G_0W_0$ @HSE is not consistent. For wide-band-gap insulators such as LiF,  $G_0W_0$ @HSE underestimates the band gap by nearly 1 eV, leading to a max error of  $-0.80$  eV.

## 2. Band structure

In Fig. 2 we plot the calculated band structures from WOT-SRSH and  $G_0W_0$ @WOT-SRSH for GaAs. Apart from a small shift, the bands are nearly identical. Additionally, the similarity of their curvature, especially near the band gap, can be seen

in the lower inset plot.  $G_0W_0$ @WOT-SRSH corrections do, however, result in a flattening of the valence bands compared to those of WOT-SRSH. This can be seen in the top figure, where the lowest valence band from WOT-SRSH is  $\approx 0.5$  eV lower than its  $G_0W_0$ @WOT-SRSH counterpart. This indicates that away from the band gap, there may be more significant differences between band structures of  $G_0W_0$ @WOT-SRSH and WOT-SRSH. In Secs. IV A 3 and IV A 4 we analyze these differences in greater detail.

## 3. Bandwidths

The calculated valence bandwidths for all compounds are reported in Table IV. Additional results from DFT functionals other than WOT-SRSH can be found in Supplemental Material Sec. S-III.B [101]. For zinc blende materials, where there is strong  $sp^3$  hybridization, the valence bandwidth is defined as the maximal energy difference between the top four (excluding spin degeneracy) valence bands. For the wurtzite and rock salt compounds, the valence bandwidth is defined as the maximal energy difference between the top three valence bands for LiF and MgO and the top six valence bands for AlN since it has twice as many atoms per unit cell. For more information on the states under consideration to compute bandwidths, see the leftmost column in Table IV. Unlike for QP band gaps, the effects of ZPR are not incorporated when comparing to experiment. Details on the DFT calculations (excluding WOT-SRSH) can be found in Supplemental Material [101]. For WOT-SRSH, the MAE and MSE are 1.08 and 0.81 eV respectively, suggesting the method tends to overestimate valence bandwidths by  $\approx 1$  eV.  $G_0W_0$ @WOT-SRSH has an MAE of 0.65 eV and an MSE of 0.18 eV, showing that  $G_0W_0$  corrections away from the band gap offer a significant improvement in accuracy relative to WOT-SRSH. Notably, the valence bandwidths for the zinc blende compounds are generally overestimated relative to experiment by both WOT-SRSH and  $G_0W_0$ @WOT-SRSH, while for the rock salt compounds studied, the valence bandwidths are, if anything, underestimated.

Moving to the well-established starting point functionals,  $G_0W_0$ @PBE computes bandwidths quite well, with an MAE of 0.59 eV and an MSE of  $-0.24$  eV. It also tends to underestimate bandwidths as its MSE suggests. For hybrids,  $G_0W_0$ @PBE0 and  $G_0W_0$ @HSE have MAEs of 0.68 and 0.64 eV and MSEs of 0.31 and 0.16 eV, respectively. Interestingly,  $G_0W_0$ @HSE and  $G_0W_0$ @WOT-SRSH have comparable levels of accuracy for bandwidths. Unlike in the case of band gaps, this similar level of accuracy persists for wide-gap insulators.

## 4. *d* band energies

For each semiconductor in our set that has elements for which *d* orbitals are explicitly treated as valence states, the *d* band position, defined as the highest *d* orbital eigenenergies relative to the valence band maxima, is reported in Table V. Additional results from DFT functionals other than WOT-SRSH can be found in Supplemental Material Sec. S-III.C [101]. As in the case of band widths, the effects of ZPR are not incorporated when comparing to experiment. For all calculations, we observe a universal underestimation of the *d*-orbital

TABLE IV. QP valence bandwidths (in eV), at the WOT-SRSH and  $G_0W_0$ @DFT level, for the various compounds and functionals under study. For zinc blende materials, the valence bandwidth is defined as the maximal energy difference between the top four (excluding spin degeneracy) valence bands. For the wurtzite and rock salt compounds, the valence bandwidth is defined as the maximal energy difference between the top three valence bands for LiF and MgO and the top six valence bands for AlN. At the bottom of the table are the MAE and MSE; all are in eV and calculated using the leftmost reported experimental values. Experimental data are obtained via XPS, angle-resolved photoemission spectroscopy (ARPES), and x-ray emission spectroscopy (XES). Due to a lack of quality data on the contributions of ZPR in these results, we do not attempt to correct for such effects in our analysis.

|      | $G_0W_0$ @ PBE | $G_0W_0$ @ PBE0 | $G_0W_0$ @ HSE | WOT-SRSH | $G_0W_0$ @ WOT-SRSH | Expt.   |
|------|----------------|-----------------|----------------|----------|---------------------|---|
| InSb | 11.30          | 11.51           | 11.32          | 11.96    | 11.32               | 11.7 <sup>a</sup> (XPS), 10.8 <sup>a</sup> (ARPES)  |
| InAs | 11.90          | 12.42           | 12.34          | 13.08    | 12.34               | 12.3 <sup>a</sup> (XPS)                             |
| Ge   | 12.82          | 13.52           | 13.30          | 14.04    | 13.26               | 12.6 <sup>a</sup> (XPS)                             |
| GaSb | 11.74          | 12.36           | 12.16          | 12.83    | 12.13               | 11.6 <sup>a</sup> (XPS), 11.64 <sup>a</sup> (ARPES) |
| Si   | 11.51          | 12.32           | 12.10          | 13.07    | 12.04               | 12.5 <sup>a</sup> (XPS)                             |
| InP  | 11.22          | 11.93           | 11.72          | 12.60    | 11.72               | 11.0 <sup>a</sup> (XPS), 11.4 <sup>a</sup> (IPES)   |
| GaAs | 12.77          | 13.45           | 13.26          | 14.09    | 13.29               | 13.8 <sup>a</sup> (XPS), 13.1 <sup>a</sup> (ARPES)  |
| AlSb | 10.67          | 11.36           | 11.14          | 12.06    | 11.20               |   |
| AlAs | 11.64          | 12.39           | 12.17          | 13.35    | 12.29               |   |
| GaP  | 12.18          | 12.92           | 12.70          | 13.70    | 12.71               | 12.5 <sup>a</sup> (ARPES)                           |
| AlP  | 11.00          | 11.83           | 11.59          | 12.75    | 11.67               |   |
| C    | 22.23          | 23.23           | 23.04          | 24.02    | 23.25               | 21 <sup>a</sup> (XPS)                               |
| AlN  | 6.55           | 6.73            | 6.69           | 6.65     | 6.75                |   |
| MgO  | 5.09           | 5.19            | 5.18           | 5.07     | 5.26                | 6.5 <sup>b</sup> (XPS), 7 <sup>b</sup> (XES)        |
| LiF  | 3.50           | 3.50            | 3.50           | 3.30     | 3.51                | 3.4 <sup>c</sup> (XPS)                              |
| MAE  | 0.59           | 0.68            | 0.64           | 1.08     | 0.65                |   |
| MSE  | -0.24          | 0.31            | 0.16           | 0.81     | 0.18                |   |

<sup>a</sup>[132].

<sup>b</sup>[133].

<sup>c</sup>[134].

locations, making the distinction between the MAE and MSE meaningless.  $G_0W_0$  corrections offer an improvement in accuracy for all starting points. For WOT-SRSH, the MSE decreases from 1.48 to 1.01 eV. For  $G_0W_0$ @PBE, it plummets from 3.8 to 1.92 eV. For  $G_0W_0$ @PBE0 it decreases from 1.7 to 1.19 eV, and for  $G_0W_0$ @HSE it decreases from 1.47 to 1.03 eV. In total,  $G_0W_0$ @HSE and  $G_0W_0$ @WOT-SRSH appear to perform the best and offer a comparable level of accuracy. However, both methods still deviate from experimental reports by  $\approx 1$  eV.

## B. Parameter sensitivity of WOT-SRSH and $G_0W_0$ @WOT-SRSH

The IP ansatz used to tune the range-separation parameter in the WOT-SRSH functional determines  $\gamma$  uniquely for a given choice of  $\alpha$  and  $\varepsilon_\infty$ . However, there can be ambiguities in the selection of  $\alpha$  and  $\varepsilon_\infty$ , with consequences for the predictive power of WOT-SRSH band gaps. Assuming first that  $\varepsilon_\infty$  has been computed accurately and that  $\beta$  is set to enforce  $\alpha + \beta = \varepsilon_\infty^{-1}$ , there exists, in principle, a range of

TABLE V. QP highest  $d$  band positions, at the WOT-SRSH and  $G_0W_0$ @DFT level, for the various functionals and  $d$ -electron containing compounds under study. At the bottom of the table are the MAE and MSE; all are in eV and measured relative to the leftmost reported experimental values. All experimental data are obtained via x-ray photoemission spectroscopy (XPS). Due to a lack of quality data on the contributions of ZPR in these results, we do not attempt to correct for such effects in our analysis.

|      | $G_0W_0$ @ PBE | $G_0W_0$ @ PBE0 | $G_0W_0$ @ HSE | WOT-SRSH | $G_0W_0$ @ WOT-SRSH | Expt.   |
|------|----------------|-----------------|----------------|----------|---------------------|---|
| InSb | 16.18          | 16.74           | 16.57          | 16.24    | 16.55               | 17.1 <sup>a</sup> , 16.98 <sup>b</sup> , 17.41 <sup>c</sup> |
| InAs | 15.31          | 16.1            | 16.04          | 15.8     | 16.03               | 16.9 <sup>a</sup> , 17.40 <sup>c</sup> , 17.38 <sup>d</sup> |
| Ge   | 26.97          | 28.32           | 28.13          | 27.25    | 28.09               | 29.4 <sup>f</sup>   |
| GaSb | 17.12          | 15.52           | 18.15          | 17.52    | 18.11               | 18.8 <sup>a</sup> , 18.9 <sup>g</sup>                       |
| InP  | 14.86          | 15.71           | 15.57          | 15.37    | 15.55               | 17.1 <sup>a</sup>   |
| GaAs | 16.81          | 17.98           | 17.81          | 17.14    | 17.83               | 18.7 <sup>a</sup> , 18.7 <sup>b</sup> , 18.82 <sup>c</sup>  |
| AlSb | 29.68          | 30.66           | 30.5           | 30.05    | 30.55               | 31.15 <sup>e</sup> , 31.60 <sup>d</sup>                     |
| AlAs | 36.66          | 37.97           | 37.82          | 37.56    | 38.03               | 39 <sup>h</sup>   |
| GaP  | 16.03          | 17.2            | 17.03          | 16.64    | 17.02               | 18.6 <sup>a</sup> , 18.7 <sup>c</sup>                       |
| MAE  | 1.92           | 1.19            | 1.03           | 1.48     | 1.01                |   |
| MSE  | -1.92          | -1.19           | -1.03          | -1.48    | -1.01               |   |

<sup>a</sup>[135]; <sup>b</sup>[136]; <sup>c</sup>[137]; <sup>d</sup>[138]; <sup>e</sup>[139]; <sup>f</sup>[140]; <sup>g</sup>[141]; <sup>h</sup>[142].

choices of  $\alpha$  for each material where one can find an optimal  $\gamma > 0$  satisfying the IP ansatz. These optimal  $(\alpha, \gamma)$  pairs produce band gaps which can differ by up to a few hundred meV. Some of the ambiguity in selecting  $\alpha$  is avoided by setting  $\alpha = 0.25$  by default, but as discussed in Sec. II A an optimal  $\gamma$  cannot always be found when  $\alpha = 0.25$ , especially if  $\varepsilon_\infty^{-1} \sim 0.25$ . In such cases,  $\alpha$  must be varied until it becomes possible to find an optimal  $\gamma$  which satisfies the IP ansatz and the constraint  $\gamma > L^{-1}$ , where  $L$  is the unit cell size used in the calculations. Additionally, it should be noted that while in principle  $\alpha$  can be increased to be as large as 1, in practice values approaching unity are generally considered to be unphysically large for most systems [68]. Thus, WOT-SRSH predictions are, in practice, more precise than those one would obtain from considering the full range of  $\alpha$  values.

Nonetheless, it is of significant interest to explore the ambiguity in selecting  $\alpha$  in the WOT-SRSH framework further and its consequences for  $G_0W_0$ @WOT-SRSH. To do so, we systematically vary  $\alpha$  and  $\gamma$  and compute GKS and  $G_0W_0$  QP band gaps for AlN. AlN is a good candidate for investigation since it has a dielectric constant that is very close to 0.25 and its band gap exhibits significant variation, on the order of hundreds of meV, between optimal  $(\alpha, \gamma)$  pairs. The difference  $\Delta E_g = E_g - E_{g,\text{ref}}$  between computed band gaps, relative to the chosen reference gaps  $E_{g,\text{ref}}$  for SRSH and  $G_0W_0$ @SRSH calculations of AlN, can be seen over a range of  $\alpha$  and  $\gamma$  in Fig. 3. Note that, as indicated, we are using, strictly speaking, the SRSH functional, as opposed to WOT-SRSH, meaning the IP ansatz is not satisfied for most of the data shown in Fig. 3. The only overall constraint applied here is  $\beta = \varepsilon_\infty^{-1} - \alpha$ . Additionally, the  $G_0W_0$  calculations presented here are slightly underconverged, using 256 bands to construct  $\varepsilon$  and  $\Sigma$ . Pairs of  $(\alpha, \gamma)$  satisfying the IP ansatz are marked with black diamonds, and the reference band gap  $E_{g,\text{ref}}$  is chosen to be the band gap obtained with the WOT-SRSH parameters of prior work [68] at either the DFT or  $G_0W_0$  level. A range of  $\pm 100$  meV about this reference value is specified in white in the colormap.

Overall,  $G_0W_0$  corrections to the SRSH starting point substantially reduce the sensitivity of the computed band gap to variations in  $\alpha$  and  $\gamma$  by about a factor of 3. Specifically, at the SRSH level  $\Delta E_g$  varies by 6.0 eV for the large ranges of  $\alpha$  and  $\gamma$  considered, while for  $G_0W_0$ @SRSH it varies by only 2.14 eV. This reduction in sensitivity becomes much more pronounced when only  $(\alpha, \gamma)$  pairs satisfying the IP ansatz are considered. At the WOT-SRSH level, the  $\Delta E_g$  values produced by these pairs have a range of 322 meV and depart from the white  $\pm 100$ -meV range about  $E_{g,\text{ref}}$  for the somewhat unphysical larger choices of  $\alpha$ . In contrast, at the  $G_0W_0$ @WOT-SRSH level, the exhibited range is only 26 meV. This reduction is by more than an order of magnitude, and substantially lower than the reduction observed for the overall SRSH functional. A similar set of trends is also observed for the other materials (see Supplemental Material Sec. S-IV.1 [101]).

We also note that there exists some uncertainty as to how to select the long-range screening  $\alpha + \beta = \varepsilon_\infty^{-1}$ . For example, one could compute  $\varepsilon_\infty$  by considering the head of the RPA dielectric matrix used in GW [5]. However, the response to a finite electric field can also be employed, leading to a beyond-RPA value of  $\varepsilon_\infty$  [143,144]. The inclusion of local

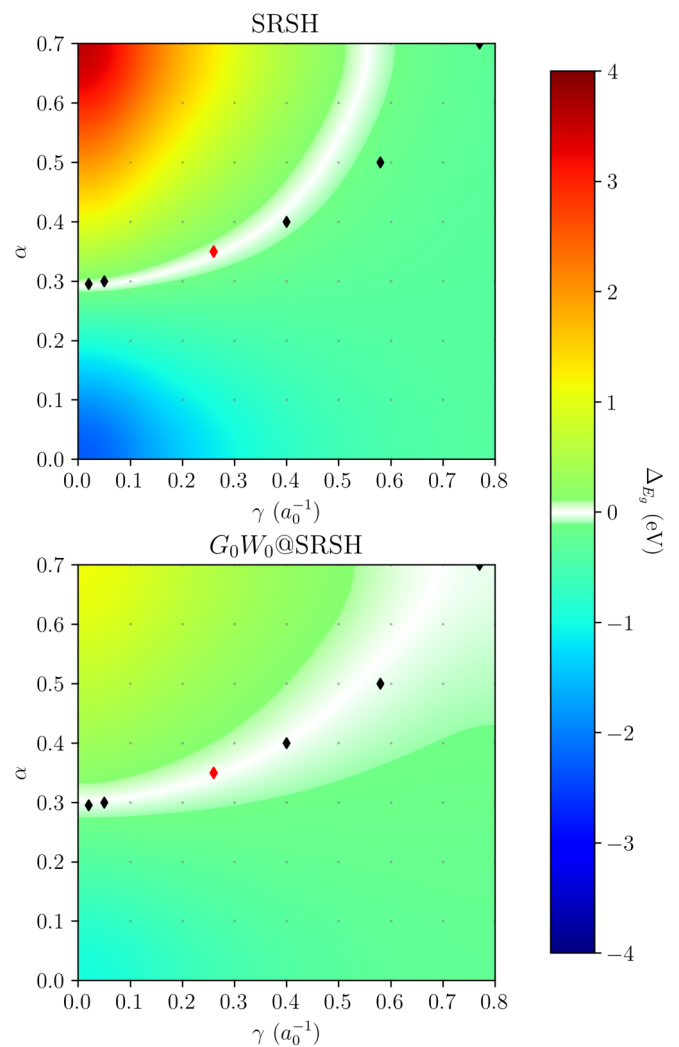


FIG. 3. The direct band gap of AlN, relative to a reference value ( $\Delta E_g = E_g - E_{g,\text{ref}}$ ) at the SRSH and  $G_0W_0$ @SRSH levels, interpolated over a wide range of  $(\gamma, \alpha)$  values. The grid of performed calculations is represented as gray dots, and the pairs satisfying the WOT-SRSH constraint are depicted as diamonds, with the reference pair for  $\Delta E_g$  in red. A range of  $\pm 100$  meV about the reference is shaded in white.  $G_0W_0$  can be seen to suppress the overall variation at the SRSH level by about a factor of 3. Moreover, the pairs satisfying the WOT-SRSH constraint can be seen to leave the  $\pm 100$ -meV range for the somewhat unphysically large values of  $\alpha$  for SRSH, but not for  $G_0W_0$ @SRSH.

field effects for such methods can also significantly affect the calculated response [145]. Additionally, the underlying DFT functional affects the computed  $\varepsilon_\infty$ . In fact, it is even possible to self-consistently update the value of  $\alpha + \beta$  in an SRSH functional so that it matches the value obtained from a calculation of the dielectric constant using said updated functional [41,146].

Accordingly, we consider the effects of varying the dielectric constant for AlN. For the WOT-SRSH functional used here,  $\varepsilon_\infty$  was computed in prior work [68] via finite electric fields using a PBE0 functional. In lieu of re-computing the optimal  $\gamma$  value for different values of  $\varepsilon_\infty$ , we simply consider the effects of changing  $\varepsilon_\infty$  by  $\pm 10\%$  while keeping



$\alpha$  and  $\gamma$  fixed. This choice likely exacerbates the sensitivity of the functional because the IP ansatz is slightly violated for the values of  $\varepsilon_\infty$  which differ from the original one used to tune  $\gamma$ . As can be seen in Supplemental Material Sec. S-IV.2 [101], these perturbations in  $\varepsilon_\infty$  result in the band gap changing by 260 meV at the SRSH level but only 80 meV at the  $G_0W_0$ @SRSH level. This behavior is in line with the approximately threefold band gap range flattening observed above. A similar trend is also observed for the other materials considered in Supplemental Material Sec. S-IV.2 [101].

## V. CONCLUSIONS

We have shown that a new WOT-SRSH class of range-separated hybrid functionals, which is tuned to satisfy an IP ansatz for localized orbitals and to have the correct asymptotic boundary conditions of the Coulomb potential in solids, offers an excellent starting point for  $G_0W_0$  calculations of the electronic structure for a wide range of semiconductors and insulators.  $G_0W_0$ @WOT-SRSH calculations perform at a level of accuracy that is equal to or better than state-of-the-art  $evGW$  and  $QSGW$  calculations at a lower computational cost. Additionally,  $G_0W_0$ @WOT-SRSH tends to improve the description of states deeper in the valence band manifold, leading to more accurate valence bandwidths and  $d$  band

locations. Finally, we have shown that  $G_0W_0$ @WOT-SRSH corrections greatly reduce the sensitivity of computed band gaps to variations in the underlying WOT-SRSH parameters that can arise from ambiguities in the optimal tuning procedure. In total, these calculations demonstrate that pairing WOT-SRSH with single-shot  $G_0W_0$  methods offers a high-accuracy predictive method for calculating QP properties of materials with a wide range of band gaps.

## ACKNOWLEDGMENTS

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