

Antiferromagnetic order in Co-doped Fe₅GeTe₂ probed by resonant magnetic x-ray scatteringXiang Chen^{1,2,*}, Enrico Schierle³, Yu He^{4,2,1}, Mayia Vranas⁵, John William Freeland⁶, Jessica L. McChesney⁶, Ramamoorthy Ramesh^{7,1,2}, Robert J. Birgeneau^{2,1,7} and Alex Frano^{5,8,†}¹Materials Science Division, Lawrence Berkeley National Lab, Berkeley, California 94720, USA²Physics Department, University of California, Berkeley, California 94720, USA³Helmholtz-Zentrum Berlin für Materialien und Energie, BESSY II, D-12489 Berlin, Germany⁴Department of Applied Physics, Yale University, New Haven, Connecticut, 06511, USA⁵Department of Physics, University of California, San Diego, California 92093, USA⁶Advanced Photon Source, Argonne National Laboratory, Argonne, Illinois 60439, USA⁷Department of Materials Science and Engineering, University of California, Berkeley, California 94720, USA⁸Canadian Institute for Advanced Research, Toronto, Ontario ON M5G 1M1, Canada

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The quasi-two-dimensional van der Waals magnet Fe_{5-δ}GeTe₂ has emerged as a promising platform for electronic and spintronic functionalities at room temperature, owing to its large ferromagnetic ordering temperature $T_C \approx 315$ K. Interestingly, by cobalt (Co) substitution of iron in F5GT, i.e., (Fe_{1-x}Co_x)_{5-δ}GeTe₂ (Co-F5GT), not only can its magnetic transition temperature be further enhanced, but the magnetic and structural ground states can also be tuned. Specifically, an antiferromagnetic (AFM) order is induced beyond the Co doping level $x \geq 0.4$. Here, we investigate the magnetic properties of a Co-F5GT single crystal at $x = 0.45(1)$, by utilizing the element-specific, resonant magnetic x-ray scattering technique. Our study reveals an A-type, Ising-like AFM ground state, with a transition temperature $T_N \approx 340$ K. In addition, our work unveils an important contribution from Co magnetic moments to the magnetic order. The application of the in-plane magnetic fields gradually polarizes the spin moments along the field direction, but without inducing incommensurate spin texture(s).

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The nature of the cleavable, quasi-two-dimensional (quasi-2D) van der Waals (vdW) materials offers rich platforms for exploring both exotic physical phenomena and technological applications [1–9]. A broad variety of intriguing physical phenomena have been reported in recent years by investigating the (atomically thin) vdW bonded compounds. Some exceptional phenomena include unconventional superconductivity in twisted graphene [10], the nonlinear Hall effect in few-layer WTe₂ [11,12], 2D magnetism in monolayer Cr₂Ge₂Te₆ [13,14], CrI₃ [15,16], and the quantum anomalous Hall effect in insulating tellurides [17,18]. Among the different materials, the layered vdW itinerant magnets represent ideal quasi-2D material systems which enable coupling between the electronic and magnetic degrees of freedom [1,4–7]. Therefore, vast opportunities arise because of the rich physical properties and the abundant possibilities for functional devices [19,20].

Very recently, some quasi-2D magnetic tellurides [21–33], such as Fe₅GeTe₂ (F5GT) and CrTe₂, with atomically thin nanoflakes for promising room temperature (RT) spintronics have been reported. The F5GT compound has a ferromagnetic (FM) transition above RT at $T_C \approx 315$ K [30,31]. Intriguingly, with cobalt (Co) substitution of iron (Fe) in F5GT [(Fe_{1-x}Co_x)_{5-δ}GeTe₂, Co-F5GT], the magnetic transition temperature is further increased up to ~ 360 K, and the

magnetic ground state switches from FM to antiferromagnetic (AFM) when the Co doping level reaches $x \geq 0.4$ [34,35]. Interestingly, a wurtzite-type polar magnetic metal, which hosts a metastable, zero-field, Néel-type skyrmion lattice at RT, was discovered at $x = 0.5$ of Co-F5GT [36,37]. The aforementioned observations of the Co-F5GT system highlight its immense tunability and capacity for unusual magnetic properties and could render applications in next-generation spintronics.

Despite the reports of exotic magnetic textures such as the chiral soliton lattices and skyrmions in F5GT or Co-F5GT [36,38,39], there is a scarcity of information from neutron scattering studies of these systems. The challenge of growing single crystals large enough for neutron experiments hinders the timely investigations of the magnetic properties of Co-F5GT, which are intriguing and display a strong dependence on the Co doping level. It is highly demanding to investigate the magnetism in Co-F5GT with a bulk sensitive scattering probe. Here, we demonstrate the applicability of utilizing the element-specific resonant magnetic x-ray scattering (RMXS) technique [40–43] to investigate the bulk magnetic properties of a specific Co-F5GT single crystal with an AFM state. By tuning the x-ray energy $E = \hbar\omega$ to either the Fe *L* or Co *L* absorption edges, the photon-excited intermediate state is sensitive to the magnetic order via the dipole allowed Fe or Co $2p \rightarrow 3d$ (*L*_{II} or *L*_{III}) transition involved in the resonant elastic scattering process [40,41]. Our study determines the contribution from both the Fe and Co spin moments to the magnetic order.

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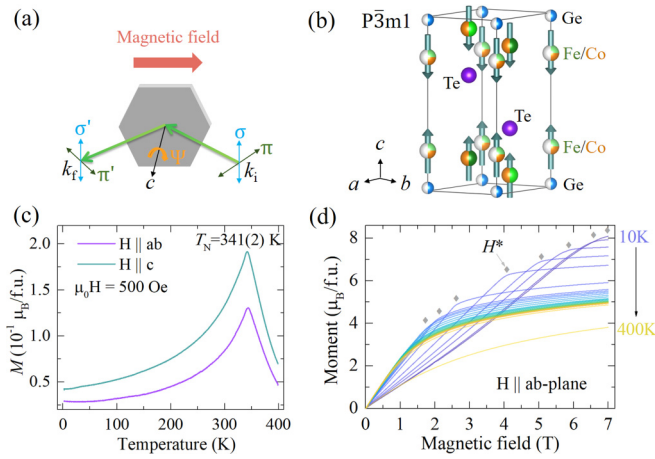


FIG. 1. (a) Schematic of the resonant magnetic x-ray scattering (RMXS) experiment on a Co-doped Fe_5GeTe_2 single crystal at $x = 0.45(1)$ [$(\text{Fe}_{0.55}\text{Co}_{0.45})_{5-\delta}\text{GeTe}_2$, Co45-F5GT]. (b) The crystal structure of Co45-F5GT with AA stacking order [34,36,37]. Colored atoms: Te (purple), Ge (blue), Co (orange). The three Fe sites are colored differently: Fe1 (light green), Fe2 (green), and Fe3 (dark green). Blank color at the Fe and Ge sites indicates half occupancy. (c) Temperature-dependent magnetization of Co45-F5GT. (d) In-plane isothermal magnetization of Co45-F5GT at varying temperatures. H^* is the saturation field, beyond which the spin moments are fully polarized along the field direction with a saturation magnetization M_{sat} .

Single crystals of Co-F5GT at $x = 0.45(1)$ (labeled here as Co45-F5GT) were synthesized using the chemical vapor transfer technique [30,31]. The chemical composition of the samples was verified by energy dispersive x-ray spectroscopy, with the cation deficiency $|\delta| \leq 0.1$. The RMXS experiments were performed at the UE46_PGM-1 beamline at Helmholtz-Zentrum Berlin (HZB) and the 29-ID IEX beamline at the Advanced Photon Source (APS) at Argonne National Laboratory. A horizontal scattering geometry was utilized with the sample lattice c direction lying within the scattering plane [Fig. 1(a)]. The Bragg peaks $\mathbf{Q} = (H \cdot \frac{2\pi}{a}, K \cdot \frac{2\pi}{b}, L \cdot \frac{2\pi}{c})$ are defined in reciprocal lattice units (r.l.u.) with lattice parameters $a = b \approx 4.02$ Å and $c \approx 9.80$ Å. The data are collected near the Fe and Co L edges to enhance the magnetic scattering signal [44–46]. The incoming x-ray is either horizontally polarized (H -pol or π -pol) or vertically polarized (V -pol or σ -pol), but the polarization of the scattered x-rays are not analyzed. Therefore, both outgoing σ' -pol and π' -pol channels contribute to the scattered intensity, which can be written as [42,43]

$$I_{\mu\nu} \propto \left| \sum_j e^{i\mathbf{Q} \cdot \mathbf{r}_j} (\mathbf{e}_\mu \times \mathbf{e}_\nu^*) \cdot \mathbf{m}_j F(E) \right|^2 \quad (1)$$

to first order in the magnetic moment \mathbf{m}_j of the ion located at site \mathbf{r}_j within the unit cell. \mathbf{e}_μ ($\mu = \sigma$ or π) and \mathbf{e}_ν ($\nu = \sigma'$ or π') are unit vectors along the polarization direction of the electric field component of the incident and outgoing x-ray beams, respectively. $F(E)$ is the nonlocal, photon energy dependent scattering tensor. Since the polarization of the scattered light

is not analyzed, the measured intensities are $I_H = I_\pi \equiv I_{\pi\sigma'} + I_{\pi\pi'}$ and $I_V = I_\sigma \equiv I_{\sigma\sigma'} + I_{\sigma\pi'}$.

The F5GT compound is composed of three identical layers with rhombohedral layer stacking (space group $R\bar{3}m$), labeled as ABC stacking [30]. It was experimentally established that the stacking order in F5GT is susceptible to external perturbations [34,36,47]. Upon replacing Fe with Co in F5GT, when $x \geq 0.4$, the crystal structure undergoes a transition from ABC stacking to AA stacking [space group $P\bar{3}m1$, as shown in Fig. 1(b)] [34]. Meanwhile, the magnetic ground state evolves from FM at $x < 0.4$ to AFM when $x \geq 0.4$. Our magnetization measurements on the Co45-F5GT sample confirm an AFM order, with $T_N = 341(2)$ K [Fig. 1(c)]. It is worth emphasizing that a type of stacking order labeled as AA' stacking was reported recently at $x = 0.5$ [36,37]. Néel-type skyrmions were identified at this doping, because of the AA' stacking order which breaks the inversion symmetry and therefore features a bulk Dzyaloshinskii–Moriya interaction [48,49]. The spin structures of Co-F5GT, however, have not yet been investigated by bulk scattering techniques. Particularly, the magnetic ground state, including the spin moment direction, shows a strong dependence on the Co doping level x [30,31,36,38]. Consequently, it is important to investigate the magnetic properties of Co-F5GT by using a direct, bulk-sensitive, element-specific scattering technique such as RMXS.

Our RMXS experiments were performed on Co45-F5GT single crystals with the AA stacking sequence [$P\bar{3}m1$, Fig. 1(b)]. A strong peak is identified at the structurally forbidden Bragg peak position $\mathbf{Q}_0 = (0, 0, 0.5)$ below the magnetic onset temperature. To understand fully the nature of this reflection, we present a thorough study of its dependence on x-ray energy, polarization, temperature, and magnetic field. Figure 2 shows the x-ray energy dependence of the \mathbf{Q}_0 reflection. Resonant peak profiles were recorded near both the Fe and Co L_{III} or L_{II} edges (Fe $L_{\text{III}} \approx 708$ eV, Fe $L_{\text{II}} \approx 720$ eV, Co $L_{\text{III}} \approx 778$ eV, and Co $L_{\text{II}} \approx 793$ eV) [44–46]. A single energy-dependent peak profile is evident near both the Fe and Co L_{III} (L_{II}) edges. This resonance in photon energy is a strong signature of the magnetic nature of the \mathbf{Q}_0 peak [42–46]. To support this argument further, the temperature dependence of the \mathbf{Q}_0 peak at the resonance energies (both Fe and Co L_{III} edges) was also examined, as shown in Figs. 2(b), 2(c) and 3. With increasing temperature, the peak intensity at either the Fe or the Co L_{III} edge becomes vanishingly small and independent of temperature above the transition temperature $T_N = 337(3)$ K (Fig. 3) [50].

Both the energy- and temperature-dependent studies in Figs. 2 and 3 demonstrate the magnetic nature of the $(0, 0, 0.5)$ peak, which is in agreement with the AFM behavior from the magnetization in Fig. 1(c). This is further supported by studying the magnetic field H dependence of the \mathbf{Q}_0 peak on resonance (at both the Fe and Co L_{III} edges), examined at select temperatures (Fig. 4). The field was applied in the ab plane of the sample [Fig. 1(a)]. With increasing magnetic field, the \mathbf{Q}_0 peak intensity is gradually suppressed and reaches almost zero when $H \geq H^*$. This can be easily understood since the applied magnetic fields are polarizing the spins along the field direction. When the magnetic moments are fully polarized under $H \geq H^*$, the magnetic structure

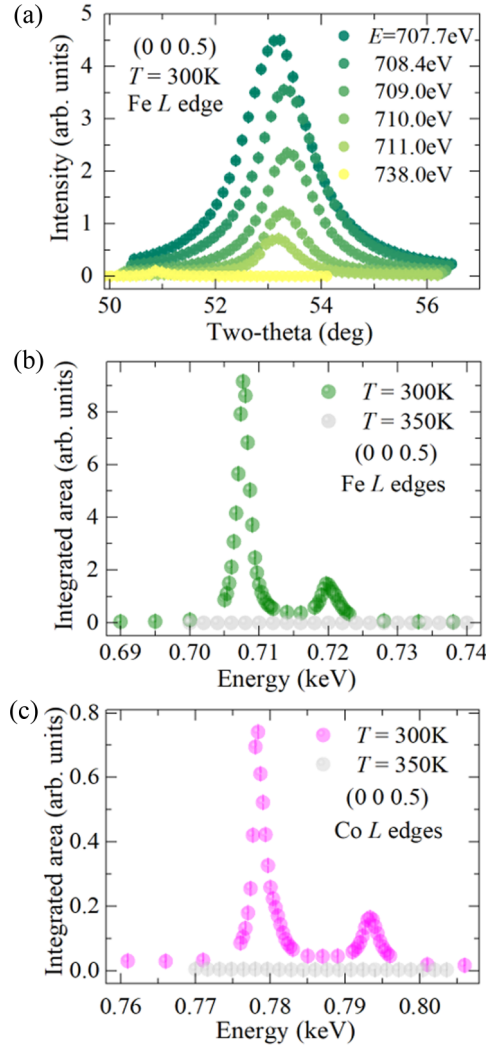


FIG. 2. Photon energy dependence of the (0, 0, 0.5) peak at $T = 300$ K or 350 K. (a) The (0, 0, 0.5) peak at $T = 300$ K near the Fe L edges. Maximum intensity is observed at the Fe L_{III} edge ($E = 707.7$ eV). (b), (c) Energy-dependent integrated area of the (0, 0, 0.5) peak at both 300 K and 350 K near the Fe L edges (b) and Co L edges (c), respectively.

enters into the polarized FM state and hence the peak intensity at \mathbf{Q}_0 disappears. By assuming that the measured peak intensity is proportional to the c component of the magnetic moment squared m_c^2 , which equals $M_{\text{sat}}^2 - M_{\text{ab}}^2$ from the magnetization data in Fig. 1(d), an excellent agreement is found between the field dependence of the resonant peak intensity at the Fe L_{III} edge and the measured magnetization squared [green line in Fig. 4(b)]. Specifically, the H^* values at different temperatures inferred from both the RMXS data and the magnetization data also match each other excellently. They are summarized and plotted as the phase diagram in Fig. 4(c).

Combining all evidence, our study confirms the AFM nature of Co45-F5GT with a propagation vector $\mathbf{Q}_0 = (0, 0, 0.5)$ and a transition temperature $T_N \approx 340$ K. The similar behavior of the magnetic peak at both the Fe and Co L edges indicates an important contribution from the Co spin moments in Co-F5GT. This explains why the saturation moment of Co45-F5GT at 10 K [Fig. 1(d)]— $M_{\text{sat}} \approx 8 \mu_B$ per formula

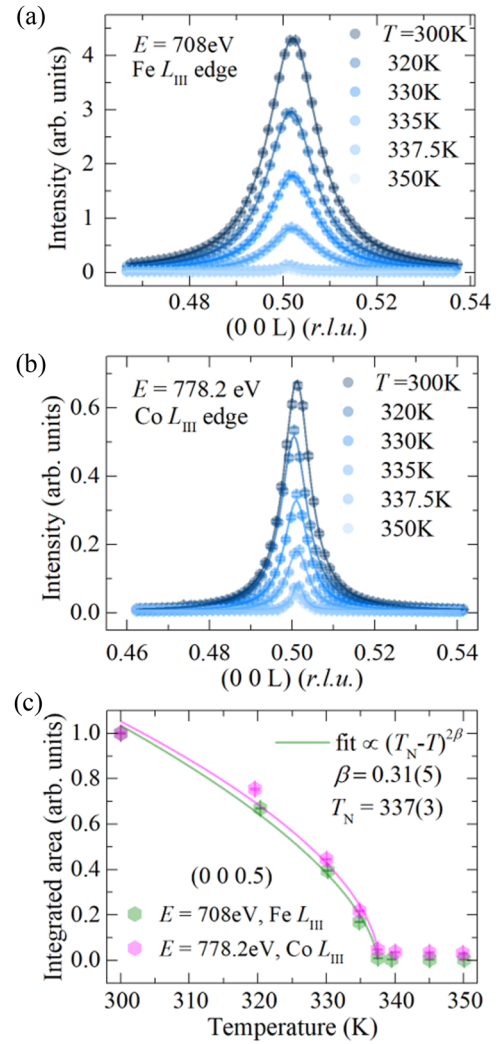


FIG. 3. Temperature dependence of the (0, 0, 0.5) peak at different energies: (a) Fe L_{III} edge, $E = 708$ eV and (b) Co L_{III} edge, $E = 778.2$ eV. Solid lines in (a) and (b) are Lorentzian fits to the peak intensity. (c) Power law fits to the integrated area of the (0, 0, 0.5) peak at both Fe and Co L_{III} edges. For better comparison, the data are normalized to 1 at $T = 300$ K.

unit (f.u.)—is only slightly smaller than the value of $M_{\text{sat}} \approx 10 \mu_B$ per f.u. in F5GT ($x = 0$). By assuming that the average Fe magnetic moment is unchanged ($\sim 2 \mu_B/\text{Fe}$) [30,31,36,47], the average Co spin moment contribution to the magnetization is estimated to be $\sim 1.1 \mu_B/\text{Co}$ in Co45-F5GT. Clearly, identifying the presence of an ordered magnetic moment at the Co site is an important result of our work.

Another important piece of information is the magnetic moment direction, specifically the magnetic structure below T_N . This can be inferred from the representational analysis [51], the photon polarization, and the azimuthal dependence of the RMXS data. The magnetic representation of the crystallographic sites of Fe or Co can be decomposed in terms of the irreducible representations (IRs) [51], with the propagation vector $\mathbf{Q}_0 = (0, 0, 0.5)$ (Table I): $\Gamma_{\text{Mag}} = 1\Gamma_2^1 + 1\Gamma_3^1 + 1\Gamma_5^2 + 1\Gamma_6^2$, where Γ_2 and Γ_3 are one-dimensional IRs with moments pointing parallel to the c axis and Γ_5 and Γ_6 are two-dimensional IRs with basis vectors lying in the ab plane.

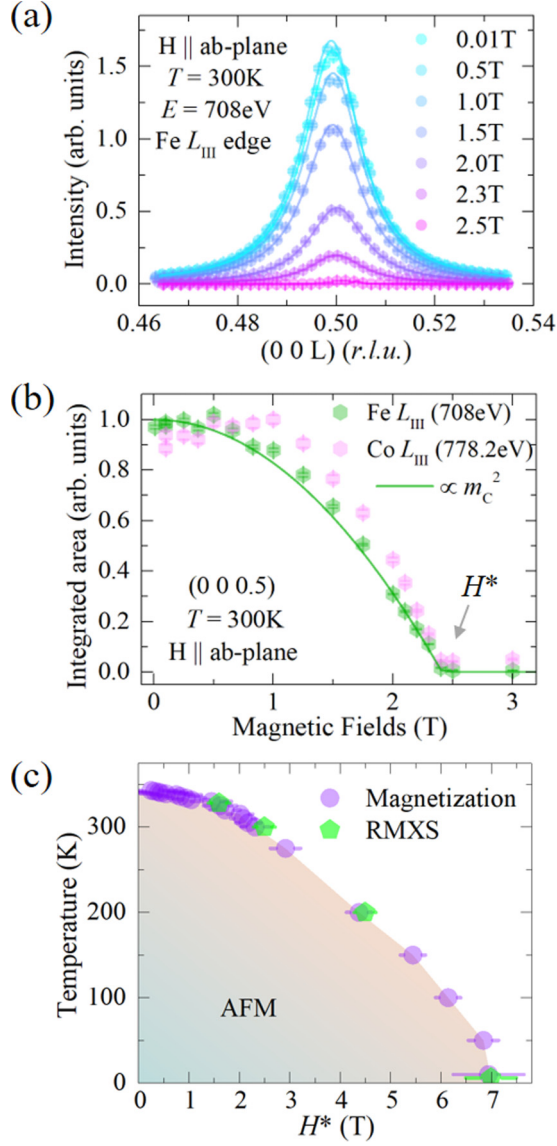


FIG. 4. Magnetic field dependence of the (0, 0, 0.5) peak on resonance. (a) The (0, 0, 0.5) peak at $T = 300$ K and $E = 708$ eV, collected under select magnetic fields. Solid lines are Lorentzian fits to the peak intensity. (b) Field dependence of the integrated area of the (0, 0, 0.5) peak at 300 K, at both Fe and Co L_{III} edges. The peak intensity is completely suppressed beyond $H^* = 2.4(1)$ T. The green line is a fit to the data by assuming the intensity is proportional to the c component of the magnetization squared m_c^2 . For better comparison, the data are normalized to 1 at zero field. (c) Phase diagram of the temperature-dependent H^* , from both the magnetization and RMXS measurements.

Considering the AFM nature, only the solutions from Γ_3 and Γ_5 are possible. Experimentally, the magnetic peak intensity at the \mathbf{Q}_0 position remains the same with either σ or π polarized incoming x-rays [Fig. 5(a)], which implies $I_\pi = I_\sigma$:

$$\left| \sum_j e^{i\mathbf{Q}\cdot\mathbf{r}_j} \mathbf{k}_f \cdot \mathbf{m}_j \right|^2 = \left| \sum_j e^{i\mathbf{Q}\cdot\mathbf{r}_j} \mathbf{k}_i \cdot \mathbf{m}_j \right|^2 + \left| \sum_j e^{i\mathbf{Q}\cdot\mathbf{r}_j} \mathbf{e}_\sigma \cdot \mathbf{m}_j \right|^2, \quad (2)$$

TABLE I. Basis vectors (BVs) for the space group $P\bar{3}m1$ with the propagation vector $\mathbf{Q}_0 = (0, 0, 0.5)$. The decomposition of the magnetic representation for the Fe1/Co1 site is $\Gamma_{\text{Mag}} = 1\Gamma_2^1 + 1\Gamma_3^1 + 1\Gamma_5^2 + 1\Gamma_6^2$. The two different atoms of the Fe1/Co1 site within the unit cell are defined: atom1, (0, 0, 0.2318); atom2, (0, 0, 0.7682). The results from the irreducible representations analysis for the other Fe/Co sites are similar.

IR	BV	Atom	BV components		
			$m_{\parallel a}$	$m_{\parallel b}$	$m_{\parallel c}$
Γ_2	ψ_1	1	0	0	1
		2	0	0	1
Γ_3	ψ_2	1	0	0	1
		2	0	0	-1
Γ_5	ψ_3	1	0	-1	0
		2	0	1	0
Γ_6	ψ_4	1	-2	-1	0
		2	2	1	0
	ψ_5	1	2	1	0
		2	2	1	0
	ψ_6	1	0	-1	0
		2	0	-1	0

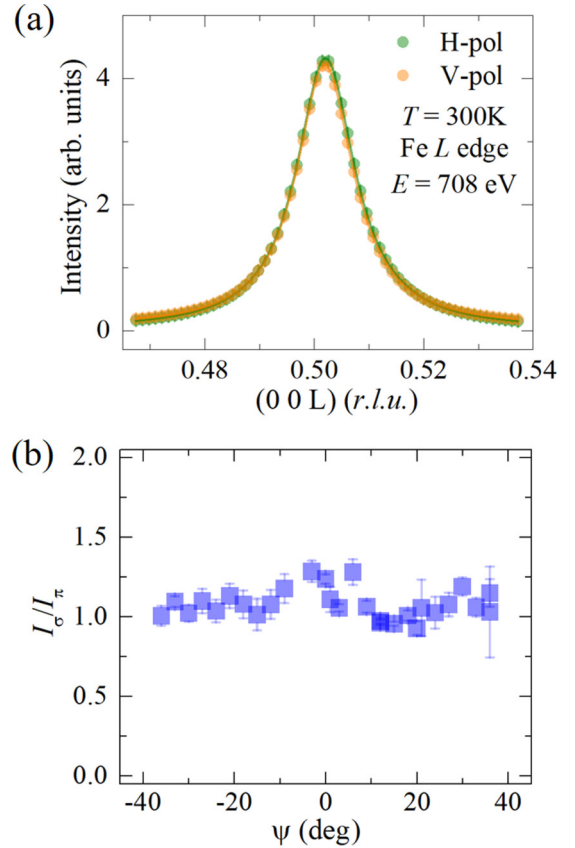


FIG. 5. (a) The (0, 0, 0.5) peak, collected at 708 eV and 300 K, with the incoming x-ray either horizontally (π -pol) or vertically (σ -pol) polarized. $I_\sigma = I_\pi$ is observed. (b) The intensity ratio $I_\sigma/I_\pi \approx 1$ in a wide range of azimuthal angle Ψ , where $\Psi = 0$ is defined when the [1 0 0] direction is parallel to the scattering plane.

where \mathbf{k}_i and \mathbf{k}_f are the unit vectors along the incoming and reflected photon wave-vector directions, respectively. The azimuthal Ψ dependence of the \mathbf{Q}_0 peak was performed by rotating the sample about the \mathbf{Q}_0 direction, which is parallel to the c axis [Fig. 1(a)], and measuring the peak intensity for both polarizations. Clearly, $I_\sigma \approx I_\pi$ is maintained over a broad range of the Ψ angle [Fig. 5(b)]. This rules out any possible contribution from Γ_5 since the presence of in-plane moments will manifest as an angle-dependent azimuthal scan by virtue of rotating the projections shown in Eq. (2). It is evident that only the solution from Γ_3 satisfies Eq. (2) ($I_\pi = I_\sigma$), regardless of the Ψ angle, since the magnet moments pointing along the c direction are parallel to the rotation axis along the \mathbf{Q}_0 direction. From these arguments, the experimental data determine an A-type AFM order with Ising moments in Co45-F5GT, as depicted in Fig. 1(b).

Under moderate magnetic fields ($H < H^*$), this AFM state is still maintained, since the (0, 0, 0.5) magnetic peak is only gradually weakened in magnitude, but without becoming incommensurate or significantly broadened in peak width. This suggests that it is unlikely to have magnetic field induced exotic spin textures in Co45-F5GT with the AA stacking order, unlike the AA'-stacked structure hosting Néel-type skyrmion lattices in Co-F5GT at $x = 0.50$ [36,37]. The contrasting magnetic textures in Co-F5GT, albeit with similar Co doping levels $x = 0.45$ and $x = 0.5$, highlight the essential role of the underlying lattice symmetry, as well as the contribution from the Co spin moments.

By tuning the photon energy to either the Fe or Co L edges, the RMXS technique measures the spin moments on the Fe and Co sites separately and reveals the contributions to the magnetism from both species. The contribution of the Co spin moments in Co-F5GT is different from the scenario in Ni-doped F5GT, in which the Ni dopants appear to be non-magnetic dilutants to the system [47]. Interestingly, both the Co- and Ni-doped F5GT systems exhibit anomalous enhancements of the magnetic transition temperature T_C . In Co-F5GT, the Co magnetic moments likely occupy the Fe1 site in a preferential manner [34,36,37], strengthening the intralayer and/or interlayer exchange couplings between the Fe and Co

magnetic moments. This might partly explain the increase of T_C in Co-F5GT. In Ni-doped F5GT, however, the enhanced three-dimensional (3D) magnetic exchange couplings might be primarily due to the structural alteration and the increase in occupancy at the Fe1 sites [47]. More experimental and theoretical investigations are needed to clarify the intriguing magnetism in both the Co- and Ni-doped F5GT systems.

In summary, our RMXS study together with the magnetization measurements on the Co45-F5GT sample confirms the Ising nature of the A-type AFM spin structure, with a propagation vector $\mathbf{Q}_0 = (0, 0, 0.5)$ and a Néel temperature $T_N \approx 340$ K. The element-specific characteristics of RMXS, observed through tuning the photon energy, highlight the sizable contribution from the Co spin moments to the AFM order. In addition, the magnetic ground state under the in-plane magnetic fields has also been investigated, suggesting the critical role of the underlying lattice symmetry for stabilizing unusual spin textures. Our work highlights the applicability of the RMXS technique to study the magnetic properties in Co-F5GT and other quasi-2D vdW magnets.

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