

1                   **CNN-based surrogate for the phase field damage model: A study on its**  
2                   **generalization across microstructure parameters for composite materials**

3                   Yuxiang Gao<sup>1</sup>, Matthew Berger<sup>2</sup>, and Ravindra Duddu<sup>3</sup>

4                   <sup>1</sup>Ph.D. student, Department of Civil and Environmental Engineering, Vanderbilt University, TN  
5                   37235(Corresponding author). ORCID: <https://orcid.org/0000-0003-0445-0685>. E-mail:  
6                   yuxiang.gao@vanderbilt.edu

7                   <sup>2</sup>Assistant Professor, Department of Computer Science, Vanderbilt University, TN 37235. E-mail:  
8                   matthew.berger@vanderbilt.edu

9                   <sup>3</sup>Associate Professor, Department of Civil and Environmental Engineering, Vanderbilt University,  
10                   TN 37235. ORCID: <https://orcid.org/0000-0002-4205-0247>. E-mail:  
11                   ravindra.duddu@vanderbilt.edu

12                   **ABSTRACT**

13                   We investigate the generalization of a CNN-based surrogate for the phase field model in pre-  
14                   dicting both damage and peak load **under uniaxial tension, given the 2D microstructure image** of a  
15                   unidirectional fiber-reinforced composite. We first discuss the phase field model and the numerical  
16                   procedure to generate training and test data from synthetic microstructures with different volume  
17                   fractions and fiber radii. We next present a two-stage approach for predicting peak load, achieved  
18                   by first transforming a given fiber-encoded microstructure image to a continuous damage field;  
19                   and second, predicting peak load from the damage field. A key finding is that the direct approach  
20                   for predicting peak load from the microstructure image using a standard regression model fails to  
21                   generalize. Instead, the damage field, even if imperfectly predicted, provides valuable cues for the  
22                   CNN in generalizing across new microstructures **within the range of parameters used in training**.  
23                   We describe several case studies to demonstrate the capability of the surrogate model to predict  
24                   damage and peak load, and to interpolate over fiber radii and volume fractions.

25 **INTRODUCTION**

26 Characterizing the mechanical performance of heterogeneous composites materials requires  
27 robust and efficient computational methods for simulating damage evolution at the microstructure  
28 scale. The phase field damage model in conjunction with the finite element method (FEM)  
29 has been widely used to predict fracture in homogeneous quasi-brittle materials; however, it is  
30 computationally expensive for heterogeneous composite material microstructures, despite the use  
31 of parallel computing. While computational multi-scale mechanics models are popular, the need  
32 for optimizing design-space parameters has led researchers to pursue computationally inexpensive  
33 surrogate models, based on micromechanics and, more recently, machine learning (ML) approaches.  
34 However, studies focusing on the generalization of ML approaches are necessary to effectively  
35 augment/replace well-established numerical methods to address complex fracture problems in a  
36 robust and efficient manner. Herein, we assess the generalization of convolutional neural network  
37 (CNN) based ML models as an efficient surrogate for the phase field damage model to predict  
38 fracture patterns in unidirectional fiber-reinforced composite microstructures.

39 Predicting the initiation and propagation of cracks (damage) in fiber-reinforced composites is  
40 challenging owing to the intrinsic variability of microstructure configurations and the nonlinear  
41 nature of damage evolution (Shakiba et al. 2019; Tan and Martínez-Pañeda 2021). While the  
42 volume fraction of fibers is well correlated with the stiffness of the composite, the distribution of  
43 fibers, fiber size, shape and misalignment affect the damage evolution and failure response (Singh  
44 and Pal 2021; Ahmadian et al. 2019). The phase field fracture/damage model has been recently  
45 used to better understand the microscale failure mechanisms in quasi-brittle composite materials  
46 in relation to microstructure parameters (Kuhn and Müller 2016; Espadas-Escalante and Isaksson  
47 2019; Zhang et al. 2019). The major advantage of the phase field damage model is that it provides  
48 a thermodynamically consistent variational framework (Francfort and Marigo 1998; Miehe et al.  
49 2010; Lo et al. 2019) to simulate fracture initiation and propagation. However, its major drawback  
50 is the computational expense arising from mesh size and time step restrictions and iterative schemes  
51 to ensure accuracy and convergence. Despite recent work implementing quasi-Newton schemes,

52 monolithic solvers, adaptive mesh refinement, domain decomposition preconditioners, length-scale  
53 insensitive formulations, continuous/discontinuous Galerkin methods, (Kristensen and Martínez-  
54 Pañeda 2020; Bharali et al. 2022; Hirshikesh et al. 2021; Svolos et al. 2020; Svolos et al. 2022; Wu  
55 and Nguyen 2018; Gupta et al. 2022), the phase field damage model is computationally expensive  
56 for fast simulation of fracture evolution in heterogeneous composite materials over size scales  
57 larger than a few millimeters, particularly in 3D.

58 In the past few years, new ideas infused into the computational fracture mechanics area inspired  
59 by the success of ML approaches in other areas of engineering, speech recognition and computer  
60 vision. Restricting our attention to ML approaches for the phase-field fracture and/or composite  
61 strength prediction, existing approaches can be classified as: physics informed (trained using the  
62 governing PDEs) and data driven (i.e. trained using only data from experiments and/or computer  
63 simulations). Physics-informed neural networks (PINNs), originally introduced by Raissi et al.  
64 (Raissi et al. 2019), are a class of coordinate-based networks that can be explicitly optimized to  
65 satisfy a governing PDE. Application of PINNs to phase field fracture simulation is quite limited,  
66 for example, Goswami et al. (Goswami et al. 2020) proposed transfer learning enhanced PINNs.  
67 Despite recent efforts to speed up, the computational expense to solve the PDEs using PINNs  
68 is comparable to that of the FEM. In contrast, the traditional data-driven surrogates utilize ML  
69 models trained on data generated by solving PDEs using other numerical methods, such as the  
70 FEM. For example, Pathan et al. (Pathan et al. 2019) presented a supervised ML model which can  
71 predict the elastic modulus and tensile strength of a fiber-reinforced composite given the image of  
72 its microstructure, using a gradient-boosted tree regression algorithm.

73 More recently, CNN-based deep learning models were developed to predict field quantities,  
74 such as stress, strain or damage, and material parameters, such as tensile strength. Croom et  
75 al. (Croom et al. 2022) built a UNet (Ronneberger et al. 2015) to predict the stress field in  
76 an additively manufactured metal directly from microstructure scans. Yang et al. (Yang et al.  
77 2021a; Yang et al. 2021b) developed a conditional GAN to predict strain and stress fields from 3D  
78 printed microstructure geometries. Bhaduri et al. (Bhaduri et al. 2021) applied UNet architecture

79 and transfer learning technique to predict the stress field in fiber-reinforced composite with a  
80 weighted mean square loss function. Mohammadzadeh et al. (Mohammadzadeh and Lejeune  
81 2021) proposed a MultiRes-WNet architecture based on convolutional blocks to predict binary  
82 damage field given an image of material distribution. Sepasdar et al. (Sepasdar et al. 2022)  
83 implemented UNet to predict stress field at the early stage of damage initiation and binary crack  
84 pattern for any given microstructure geometry, using a modified mean average error as the loss  
85 function. Zhou et al. (Zhou et al. 2021) used a CNN-based model for composite strength prediction  
86 given the microstructure RVE. However, to best of our knowledge, there are no studies focusing on  
87 surrogates for the phase field damage model and their generalization.

88 In this article, we present a fast CNN-based surrogate for the phase field damage model, which  
89 belongs to the class of data driven deep learning approaches. Based on our observation that CNNs  
90 poorly generalize for predicting peak load (i.e., material strength) if provided with only the 2D  
91 microstructure (e.g., binary image that encodes the matrix and fibers), we propose a two-stage  
92 approach for predicting peak load. In first stage, we translate the microstructure image to its  
93 corresponding damage field at peak load, and in the second stage, we predict the peak load from  
94 the damage field. We find that this two-stage approach generalizes better compared to the one-  
95 stage or direct approach that predicts peak load exclusively from the microstructure image. To  
96 provide guidance on the selection of training sets for future applications, we further investigate  
97 the generalization of our CNN-based surrogate model to predict damage fields and peak loads for  
98 microstructures with different volume fractions and fiber radii.

## 99 METHODOLOGY

### 100 Phase field damage model

101 The phase-field damage model employed here is an extension of Miehe's model (Miehe et al.  
102 2010) proposed by Lo et al. (Lo et al. 2019). This model properly accounts for tension-compression  
103 asymmetry of damage evolution in brittle materials. To describe the model formulation, we consider  
104 an arbitrary linear elastic solid  $\Omega \subset \mathbb{R}^2$  with external boundary  $\partial\Omega$ , as shown in Fig. 1. The boundary  
105  $\partial\Omega$  is split into disjoint Dirichlet and Neumann boundaries denoted by  $\partial\Omega_D$  and  $\partial\Omega_N$ , respectively.

106 The fiber domain  $\Omega^{(f)}$  is assumed to be undamaged at all times. The damaged state at any position  $\mathbf{x}$   
 107 and time  $t$  in the matrix domain  $\Omega/\Omega^{(f)}$  is described by a scalar phase-field variable  $D(\mathbf{x}, t) \in [0, 1]$ ,  
 108 where  $D = 0$  represents undamaged state and  $D = 1$  the fully damaged state.

109 In the hybrid phase field model formulation, there are two unknown field variables that need to  
 110 be solved for: the displacement field  $\mathbf{u}$  and the phase-field damage variable  $D$ . To determine the  
 111 displacement field  $\mathbf{u}$ , we solve the governing equations of the elastostatic boundary value problem  
 112 along with the corresponding boundary conditions over a space-time domain  $Q = \Omega \times [0, T]$ :

$$113 \quad \begin{cases} \nabla \cdot \boldsymbol{\sigma} = \mathbf{0} & \text{in } Q \\ \mathbf{u} = \mathbf{u}^* & \text{on } \partial\Omega_D \times [0, T] \\ \boldsymbol{\sigma} \cdot \mathbf{n} = \mathbf{t}^* & \text{on } \partial\Omega_N \times [0, T] \end{cases} \quad (1)$$

114 where  $\mathbf{u}^*$  represents a prescribed displacement on the Dirichlet boundary  $\partial\Omega_D$ ,  $\mathbf{t}^*$  represents a  
 115 prescribed traction on the Neumann boundary  $\partial\Omega_N$ ,  $\boldsymbol{\sigma}$  is the Cauchy stress tensor defined by the  
 116 isotropic linear elastic constitutive law under the plane strain assumption, and  $\mathbf{n}$  is the outward  
 117 normal to the boundary  $\partial\Omega_N$ . To determine the damage field  $D$ , we solve the governing equation  
 118 for phase-field evolution in the rate form along with the associated boundary conditions:

$$119 \quad \begin{cases} \eta \dot{D} = l_c \Delta D - \frac{D}{l_c} + 2(1 - D) \frac{H}{G_c} & \text{in } Q \\ \nabla D \cdot \mathbf{n} = 0 & \text{on } \partial\Omega \times [0, T] \end{cases} \quad (2)$$

120 where  $\Delta$  is the Laplace operator,  $\eta$  is the viscous regularization parameter,  $H$  is the history field  
 121 variable ensuring the monotonic increase of the phase-field variable defined as (Lo et al. 2019)

$$122 \quad H(\mathbf{x}, t) = \max_{\tau \in [0, t]} [\psi^+(\boldsymbol{\varepsilon}(\mathbf{x}, \tau))], \quad (3)$$

123  $\boldsymbol{\varepsilon}$  is the small strain tensor, and  $\tau$  is a pseudo-time variable related to the applied load/displacement  
 124 rate. In the above equation, by defining the history field variable equal to the maximum value of

125 the tensile portion of the strain energy  $\psi^+(\boldsymbol{\varepsilon})$  we ensure that damage is irreversible. We use the  
126 standard Galerkin finite element procedure to discretize and solve the equations using a staggered  
127 approach in the open source finite element software FEniCS (Alnæs et al. 2015). Full details of the  
128 procedure to solve these equations can be found in our recent article (Sun et al. 2021).

## 129 **Ground truth data generation by finite element method**

130 We consider an archetype problem of damage/fracture evolution within the 2-D representative  
131 volume element (RVE) of a unidirectional fiber-reinforced composite under applied uniaxial tension  
132 in plane strain. We generated several 2-D geometries of using the L-BFGS-B algorithm described  
133 in Section 2.3 of Pathan et al. (Pathan et al. 2017). Specifically, we chose square-shaped RVEs  
134 with side length of  $L = 50 \mu\text{m}$  and varied two microstructure parameters: the volume fraction  $V_f$  is  
135 varied from 0.2 to 0.5, and the fiber radii  $R_f$  is chosen as  $2 \mu\text{m}$ ,  $4 \mu\text{m}$  or  $6 \mu\text{m}$ . To prevent damage  
136 artifacts from initiating and propagating near fibers that are near or intersect with the external  
137 boundary, we limit the fiber positions to a square interior region with side length  $L - (R_f + 0.04L)$ ,  
138 where  $R_f$  is the chosen fiber radii and  $0.4L$  is a small offset determined by trial-and-error. We  
139 assume that the fiber is a linear elastic material with a large stiffness (the Young's modulus is 74  
140 GPa and the Poisson's ratio is 0.2); whereas the matrix is a linear elastic material with a small  
141 stiffness (the Young's modulus is 3.5 GPa and the Poisson's ratio is 0.35) and accrues damage  
142 (critical strain energy release rate is  $10 \text{ J/m}^2$ ). For simplicity, the fiber and the matrix interface are  
143 assumed to be perfectly bonded, so failure by debonding does not occur. We take the phase-field  
144 viscosity (stabilization) parameter  $\eta = 1.2$  and length scale parameter  $l_c = 0.8 \mu\text{m}$ .

145 The RVEs were discretized with a triangular finite element mesh with edge length  $\approx 0.1 \mu\text{m}$ ,  
146 using the open-source mesh generation software Gmsh (Geuzaine and Remacle 2009). As illustrated  
147 in Fig. 1, we enforce the traction free condition on the left and right boundary of the RVE, and free  
148 slip (or roller) condition on the bottom boundary. The bottom left corner is pinned to prevent rigid  
149 body motion. At each pseudo-time step, to simulate the uniaxial extension test we apply a uniform  
150 displacement increment of  $0.01 \mu\text{m}$  on the top boundary. By summing the nodal reaction forces  
151 on the top boundary we obtain the applied load at each time step. We stop the simulation when the

152 applied load drops to 15 % of the maximum/peak load. By dividing the peak load by the length of  
153 the RVE, we can determine the tensile strength, although we prefer to compare peak load values  
154 for different cases. The peak load and the corresponding 2D damage field image obtained from the  
155 finite element simulations are stored and used as the ground truth data for training and testing the  
156 ML models, as discussed below.

157 As shown in Table 1, we generated five datasets with different volume fractions and fiber radii  
158 for training and evaluating the proposed ML model. For dataset A, as a baseline, we consider 800  
159 samples in the training set and 200 samples in the test set. All samples in dataset A have the same  
160 fiber radius of  $4 \mu\text{m}$  and volume fraction of 0.5. The volume fraction of the samples in dataset B  
161 and C are set to be 0.5. In the training set of dataset B, we have 2000 samples (1000 each) with  
162 fiber radii of 2 or  $6 \mu\text{m}$ ; whereas the training set of dataset C contains 2001 samples (667 each)  
163 with fiber radii of 2, 4, and  $6 \mu\text{m}$ . The test sets of datasets B and C are the same as that of dataset  
164 A. The fiber radius of dataset D and E samples is set to be  $4 \mu\text{m}$ . There are 2000 samples (1000  
165 each) in the training set of dataset D with volume fractions of 0.2 or 0.5; whereas the training set of  
166 dataset E has 2080 samples (130 each) with a range of volume fractions from 0.2 to 0.5. The test  
167 set of the datasets D and E are the same, which has 200 samples with a volume fraction of 0.3 and  
168 fiber radius of  $4 \mu\text{m}$ .

#### 169 **Baseline model: Predicting peak load from microstructure**

170 We first consider a simple ML approach that can directly predict the peak load given the  
171 rasterized image of the RVE as input, as illustrated in Fig. 2a. The architecture of this CNN-  
172 based regression model is based on that proposed by Simonyan et al. (Simonyan and Zisserman  
173 2015). The RVE image is converted to a  $512 \times 512$  pixel binary image, where fibers are marked  
174 as 1 and matrix is marked as 0. The  $1 \times 512 \times 512$  pixel input image is fed through a series of  
175 convolutional blocks, where each block performs: (1) a convolution with a set of learnable filters,  
176 in turn producing a multi-channel image; (2) batch normalization (Ioffe and Szegedy 2015); (3) a  
177 leaky ReLU activation (Khalid et al. 2020); and (4) spatial downsampling of the image by a factor  
178 of 2. We apply this series of operations until we reach a 512-channel image with spatial resolution

179 2  $\times$  2. This image is then flattened into a 2,048-dimensional vector, and through a 2-layer multilayer  
180 perceptron (MLP), the model produces a single value as output. This model is trained to minimize  
181 the mean squared error (MSE) between the microstructure’s ground-truth and predicted peak load.

182 To evaluate the accuracy of this baseline or one-stage ML model in predicting the peak load, we  
183 trained and tested it on a sample dataset with fixed microstructure design parameters, that is, with a  
184 fiber radius of 4  $\mu\text{m}$  and a volume fraction of 0.5. The dataset includes 1000 paired RVE geometry  
185 images and the corresponding peak load values, and we split it into 80 % training set and 20 % test  
186 set. As evident from Fig. 2, this direct approach overfits to the training data (c.f. Fig. 2b), and fails  
187 to generalize to novel microstructures not observed during training (c.f. Fig. 2c). This suggests  
188 that microstructure geometry alone does not provide sufficient cues for this baseline CNN model  
189 to properly reason about the microstructure’s peak load. These results motivate our two-stage  
190 approach, where we seek to provide a CNN-based peak load predictor with richer information as  
191 input, in the form of a damage field image.

## 192 **CNN-based image-to-image-to-value surrogate model**

193 To obtain an improved prediction of peak load, we now consider a two-stage ML approach that  
194 first predicts the damage state given the rasterized image of the RVE as input, and then uses the  
195 damage field image to predict the peak load. We hypothesize the damage field at peak load can  
196 provide better information for peak load prediction than just the RVE geometry. This is based on  
197 the rationale that the phase field damage includes information on the strain energy state, material  
198 fracture toughness, along with the RVE geometry. Because this two-stage approach can predict  
199 both the damage field and peak load, it can be used as a surrogate for the phase field damage model.

200 To predict damage field image from the RVE image, we employed a CNN-based image-to-image  
201 model with an encoder-decoder and skip connection architecture, similar to the UNet ([Ronneberger  
et al. 2015](#)), as shown in Fig. 3a. The encoder (left side) accepts the RVE image as input, a  
202 single-channel binary encoding of the fibers (c.f. Sec. Baseline model: Predicting peak load from  
203 microstructure). The purpose of the encoder is to produce a series of multi-channel images that  
204 increase in channel resolution, and decrease in spatial resolution, using a learned set of convolutional  
205

206 blocks. The decoder (right side) accepts as input the last multi-channel image produced by the  
 207 encoder, and performs a series of learnable convolutional blocks, each interleaved with spatial  
 208 upsampling, until we reach a target image resolution of  $512 \times 512$  pixels – it is this image that we  
 209 treat as the predicted damage field. Skip connections are utilized, wherein multi-channel images  
 210 produced by the encoder are concatenated, channel-wise, onto the images produced by the decoder,  
 211 limited to layers that have equivalent spatial resolution.

212 The parameters of the model are found by minimizing a mean absolute error (MAE) objective  
 213 between the predicted damage field and the ground-truth damage field. Because the ground-truth  
 214 damage fields are obtained from FEM-based simulations, each field is defined over a triangular  
 215 mesh, thus incompatible with the model’s predicted image represented over a regular square grid.  
 216 To resolve this issue, we resample the FEM damage field onto a  $512 \times 512$  regular grid, which  
 217 enables us to compute a pixel-wise MAE loss for optimization. Specifically, for each vertex on  
 218 the triangular finite element mesh, we find the cell in the regular grid that contains this vertex and  
 219 contribute this vertex’s nodal value to each of the grid cell’s four vertices. The contribution is  
 220 obtained through a bilinear interpolation weight  $w$  within each cell to calculate the grid point value  
 221  $Q$ , as described in Algorithm 1

---

**Algorithm 1** Resampling from FEM results to a rasterized image based on bilinear interpolation

---

**Require:**  $M \times M$  image resolution,  $(\mathbf{X}^{L \times 1}, \mathbf{Y}^{L \times 1}) \in [0, 1] \times [0, 1]$  (Normalized finite element  
 nodal coordinates),  $\mathbf{F}^{L \times 1}$  (Nodal values of the phase field damage).

- 1: Initialize  $\mathbf{Q}^{M \times M} = \mathbf{0}$ ,  $\mathbf{W}^{M \times M} = \mathbf{0}$
- 2:  $\mathbf{X} \leftarrow \mathbf{X} \times (M - 1)$ ,  $\mathbf{Y} \leftarrow \mathbf{Y} \times (M - 1)$
- 3:  $\hat{\mathbf{X}} \leftarrow \lfloor \mathbf{X} \rfloor$ ,  $\hat{\mathbf{Y}} \leftarrow \lfloor \mathbf{Y} \rfloor$
- 4: **for**  $p \leftarrow 1$  to  $L$  **do**
- 5:   **for**  $i \leftarrow \hat{X}_p$  to  $\hat{X}_p + 1$  **do**
- 6:     **for**  $j \leftarrow \hat{Y}_p$  to  $\hat{Y}_p + 1$  **do**
- 7:        $w \leftarrow (1 - |i - X_p|)(1 - |j - Y_p|)$
- 8:        $Q_{ij} \leftarrow Q_{ij} + w f_p$
- 9:        $W_{ij} \leftarrow W_{ij} + w$
- 10:     **end for**
- 11:   **end for**
- 12: **end for**
- 13: Output the resampled image  $\mathbf{Q} \leftarrow \mathbf{Q}/(\mathbf{W} + 10^{-12})$

---

222 We employ this CNN-based surrogate model with the architecture shown in Fig. 3(a) due to the  
223 following considerations. First, encoder-decoder architectures have modest memory requirements  
224 and are computationally efficient; this is especially advantageous for models tasked with transform-  
225 ing one high-resolution image to another high-resolution image. Second, skip connections allow  
226 us to exploit intricate relationships between the spatial distribution of fibers and the damage field.  
227 Damage evolution in the microstructure is a *localized* phenomenon that directly depends on the  
228 stress/strain distribution in the RVE. By designing the decoder (i.e. damage field) to condition on  
229 features produced at multiple resolutions by the encoder (i.e. RVE geometry), we allow for the  
230 model to reason damage localization based on fiber distribution, at different spatial resolutions.

231 Thus, the proposed two-stage approach consists of a pair of CNN-based deep learning models:  
232 (1) a CNN tasked with predicting the phase field damage, given a microstructure image; and (2) a  
233 CNN tasked with predicting the peak load, given a damage field as an image. We opt to use the  
234 same CNN model for predicting peak load given an image, as described in Section Baseline model:  
235 Predicting peak load from microstructure. We optimize for each of these CNN models separately,  
236 wherein the damage field model is optimized solely on instances of microstructure and ground-truth  
237 damage field images (c.f. Fig. 3b); whereas the peak load model is optimized on instances of the  
238 predicted damage field and corresponding peak load value (c.f. Fig. 3c). At inference time, given  
239 a microstructure image, we first predict the corresponding damage field, and then use this damage  
240 field to predict the peak load.

## 241 **Error metrics**

242 To determine the difference between the ground truth and the CNN-based ML model predictions,  
243 we utilize two commonly used error metrics, namely root mean square relative error (RMSRE) and  
244 normalized root mean square error (NRMSE).

245 Because the peak load for each RVE is a unique value (dependent on the spatial distribution  
246 of fibers), for comparing the error across different RVEs it is more appropriate to first evaluate the  
247 relative error for each RVE and then calculate its root mean square. Hence for a given number  $N$

248 of RVEs, we define the RMSRE between ground truth  $y$  and ML model predictions  $\bar{y}$  as

249

$$\text{RMSRE} = \sqrt{\frac{1}{N} \cdot \sum_{i=1}^N \left( \frac{y_i - \bar{y}_i}{y_i} \right)^2}. \quad (4)$$

250 The damage field image is also uniquely determined by the spatial distribution of fibers, with  
251 zero damage within the fibers (i.e. fiber is linear elastic). However, the RMSRE is not suitable  
252 for evaluating the error in damage, because it can be zero at various locations within a given RVE,  
253 which means the relative error is not well defined. Therefore, for comparing the relative error  
254 of rasterized images of phase field damage between different RVEs, we define the NRMSE with  
255 respect to the RMS of the ground truth. Thus, for a ground truth image  $Y$  and a predicted image  $\bar{Y}$   
256 with  $N$  pixels, the NRMSE is defined as

257

$$\text{NRMSE} = \sqrt{\frac{\sum_{i=1}^N (Y_i - \bar{Y}_i)^2}{\sum_{i=1}^N Y_i^2}}. \quad (5)$$

258 Note that the average magnitude of the phase-field damage (i.e. the integral of damage over the  
259 RVE divided by its area) is different for each RVE. Therfore, to appropriately compare the error  
260 across different RVEs, we must use the NRMSE as defined above.

261 **RESULTS AND DISCUSSION**

262 **Preliminary check: damage field and peak load prediction**

263 We first test whether the CNN-based surrogate model is able to predict the damage field and peak  
264 load with reasonable accuracy, if only the spatial distribution of the fibers is varied. Therefore, we  
265 trained and tested the model on a sample dataset with fixed microstructure parameters of volume  
266 fraction 0.5 and fiber radius 4  $\mu\text{m}$ . This dataset includes 1000 paired binary images of RVE  
267 geometry and damage field. We split the dataset into 80 % training set and 20 % test set.

268 Figs. 4a and 4b show the model and ground truth damage field predictions and the difference for  
269 the smallest and largest NRMSE in the entire test dataset, respectively. In Fig. 4c we find that the  
270 NRMSE of more than 80 % cases is smaller than 0.17, and the average NRMSE across all cases is

0.1289, suggesting good generalization behavior on novel microstructures. As evident in Fig. 4b, larger error tends to manifest as a difference in the magnitude of damage. Otherwise, the locations in the microstructure where damage occurs, and where damage is absent, are generally preserved well in the predicted field. This suggests that the damage field input (provided to the CNN model for predicting peak load) will contain useful image cues, even when the predicted input is imperfect.

Fig. 4e shows the comparison between the peak load results obtained from the FEM and the image-to-image-to-value (surrogate) model. The data points are distributed along the diagonal, which indicates that the regression model is able to learn/capture the relationship between the damage field and the peak load. Additionally, we notice that the surrogate model tends to underestimate the peak load value at the higher end and overestimate at the lower end. This is possibly due to the relatively smaller amount of training data corresponding to the peak load extremities. The baseline model (cf. Section Baseline model: Predicting peak load from microstructure) results are shown in Fig. 4f for comparison. Fig. 4d shows that the surrogate model has more cases within 2% relative error and smaller maximum relative error for peak load prediction than the baseline model. Quantitatively, the RMSRE between surrogate model prediction and ground truth of the peak load is 0.0212, and the maximum relative error is 0.0576, which are both smaller than the baseline model (RMSRE = 0.0336, Max relative error = 0.0941). Thus, the surrogate model demonstrates an improved capability for predicting peak load compared to the baseline model.

## Model generalization and training set selection

Volume fraction and fiber radii are two microstructure design parameters that can be altered to improve the mechanical behavior of composite materials. In the above two sections, we only examined the accuracy of the surrogate model when testing and training on the dataset with the constant volume fraction and fiber radius. In this section, we address the generalization of the surrogate model across a range of volume fraction and fiber radii, which can be a formidable challenge. Particularly, it is useful to know how to select the appropriate training dataset so that the ML model performs well across different test sets spanning the range of microstructure parameters. We consider two relatively straightforward training strategies:

298 1. Extreme: train only using the dataset corresponding to the minimum and maximum values  
299 of microstructure parameters;  
300 2. Range: train using the dataset covering the entire range of microstructure parameters.

301 For both strategies, we assess the accuracy of damaged field and peak load prediction using NRMSE  
302 and RMSRE metrics, respectively. We utilize a separate dataset within the range of parameters to  
303 test the ML model's ability to interpolate or generalize.

304 *Training with the extremes of fiber radius*

305 We trained the surrogate model with the datasets (consisting of RVE and damage field images  
306 and peak load values) corresponding to volume fraction  $v_f = 0.5$  and the extremes of fiber radii  
307  $R_f = 2 \mu\text{m}$  and  $6 \mu\text{m}$ . Fig. 5(a) and Fig. 6(a) shows the model performance on the test dataset  
308 with  $v_f = 0.5$  and intermediate fiber radius  $R_f = 4 \mu\text{m}$ . The surrogate model predicted the  
309 locations of damage initiation and the overall damage pattern reasonably well (results not shown  
310 here), despite the discrepancy in the magnitude of the damage field. The NRMSE of 96 % cases  
311 are smaller than 0.23 with an average NRMSE of 0.1589 for the entire distribution, as shown in  
312 Fig. 5(a). The relative error between the peak load predictions from the surrogate model and the  
313 finite element method (FEM) is quite small with RMSRE = 0.0262, which is comparable to that  
314 from our preliminary check (RMSRE = 0.0162) in Section Preliminary check: damage field and  
315 peak load prediction. Considering that the surrogate model was never trained on fiber radii of  
316  $R_f = 4 \mu\text{m}$ , this study suggests that the two-stage approach can successfully generalize over the  
317 range of microstructure design parameters.

318 *Training with a range of fiber radii*

319 We next trained the surrogate model with the datasets of all three fiber radii  $R_f = 2, 4$  and  $6 \mu\text{m}$   
320 and tested its performance for each fiber radii, separately. The purpose of this study is understand  
321 which of the two training strategies (i.e. extreme or range) improves accuracy of the damage field  
322 and peak load prediction. Figs. 5b–5d and 6b–6d show the results assessing the surrogate model  
323 performance on the test datasets of three different fiber radii. The surrogate model is able to predict

324 damage pattern reasonably well (results not shown here), even for the cases with the largest NRMSE  
325 in the respective test datasets. However, there are magnitude differences in the damage field images  
326 corresponding to surrogate model and ground truth.

327 Figures 5b–5d show that the NMRSE of the predicted damage field is less than 0.15 for more  
328 than 90% of cases across the three test sets. We notice that the surrogate model gives the smallest  
329 average NRMSE for damage field prediction on the test dataset with radius 2  $\mu\text{m}$  (Fig. 5b). We  
330 suspect this is because the RVE contains a larger number of fibers distributed uniformly through the  
331 matrix, which reduces the variability or uncertainty arising from the spatial distribution of fibers.

332 Figures 6b–6d show that the surrogate model has more cases with < 2% relative error, demon-  
333 strating that it can generalize well when training with a range of fiber radii. Thus, the surrogate  
334 model is able to learn/capture the relationship between the damage field and the peak load.

335 *Model performance with different datasets*

336 We now examine the surrogate model’s performance across the three different datasets A, B,  
337 and C, as shown in Table 1. To elaborate our aim here: in Section Preliminary check: damage  
338 field and peak load prediction, we trained and tested the surrogate model using the dataset A with  
339 volume fracture  $v_f = 0.5$  and fiber radii,  $R_f = 4 \mu\text{m}$ , with an 80 % – 20 % split (see Table 1); in  
340 Section Model generalization and training set selection thus far, we trained and tested the surrogate  
341 model with datasets B and C, where we trained using the dataset with the extremes and range of the  
342 fiber radius, and tested using the dataset with an intermediate fiber radii. Thus, in all three datasets  
343 the test set is the same, which allows us to identify the best training strategy for the surrogate model.

344 From Table 2, it is evident that the lowest NRMSE error for damage field prediction and RMSRE  
345 for peak load prediction is obtained with dataset C, compared to datasets A and B. Thus, training  
346 the surrogate model with datasets of three different fiber radii improves its accuracy compared to  
347 training with the same fiber radius as the test dataset. Therefore, training the surrogate model with  
348 the range of fiber radii is a better choice for ensuring generalization.

349 *Training with the extremes of volume fraction*

350 We next trained the surrogate model with the datasets corresponding to fiber radius  $R_f = 4 \mu\text{m}$   
351 and the extremes of fiber volume fraction  $v_f = 0.2$  and  $0.5$ . Fig. 5e and 6e show the model  
352 performance on the test dataset with  $v_f = 0.3$  and  $R_f = 4 \mu\text{m}$ . This training and test dataset is  
353 denoted as dataset D in Table 1. The NRMSE of 85% of cases is smaller than 0.2, despite the  
354 maximum NMRSE being 0.35, as shown in Fig. 5e. Fig. 6e shows the relative errors between the  
355 peak load predictions from the surrogate model and the FEM are less than 4 % in most cases. This  
356 study illustrates that the surrogate model trained over the extremes of fiber volume fraction can be  
357 reasonably predictive at intermediate volume fractions.

358 *Training with the range of volume fraction*

359 We now investigate whether the model will exhibit improved generalization when training with  
360 the range of volume fraction. Therefore, we trained the surrogate model on a dataset consisting  
361 of 16 different volume fractions ranging from 20 % to 50 %. The fiber radius  $R_f = 4 \mu\text{m}$  is kept  
362 constant, so each additional fiber in the RVE increases the fiber volume fraction by 2%. To evaluate  
363 the accuracy of the two training strategies (i.e. extreme versus range), the surrogate model was  
364 tested on the same dataset with  $V_f = 0.3$  and  $R_f = 4 \mu\text{m}$ , as in the previous section. The details of  
365 the two relevant datasets are given in Table. 1.

366 Fig. 5f and 6f shows the results assessing the surrogate model's performance when trained and  
367 tested with dataset E spanning the entire range of fiber volume fractions. The average NRMSE is  
368 0.1481 when trained with dataset E, as opposed to 0.1477 when trained with dataset D (see Table  
369 2). The RMSRE of peak load prediction for dataset E is 2.67 %, compared to 3.27 % smaller than  
370 with dataset D.

371 This study showed that training with a range of volume fractions slightly improves the surrogate  
372 model's capability to predict peak load, compared to training with the extremes of volume fractions.  
373 Interestingly, the improvement in peak load prediction is not accompanied by an improvement in  
374 predicting the damage field (c.f. Fig. 5e and Fig. 5f). This finding indicates that a diverse coverage  
375 of volume fraction is of more benefit for peak load generalization, and helps to mitigate the

376 overprediction bias phenomena previously.

377 *Discussion*

378 In all the above studies, we restricted our attention to “generalization” across microstructure  
379 parameters within the range used in training and to loading/boundary conditions corresponding to  
380 uniaxial tension under plane strain. To assess generalization outside of the parameter range, we  
381 performed a preliminary study by training the model on the dataset with  $R_f = 2 \text{ \mu m}$  and testing  
382 on the dataset with  $R_f = 6 \text{ \mu m}$ . We find that the peak load error from the two-stage CNN model  
383 is larger for prediction outside the parameter range (NMRSE = 0.0542) compared to prediction  
384 within the parameter range (NMRSE = 0.0262). However, if we include a few RVEs  $R_f = 6 \text{ \mu m}$  in  
385 the training dataset, we find that two-stage ML model accuracy improves for prediction outside the  
386 parameter range (NMRSE = 0.0322). This is consistent with our observation that the error is the  
387 smallest when trained with the range of fiber radii or volume fraction. These results are included in  
388 Figure S6 in the supplementary material. Model generalization for the loading/BCs can be achieved  
389 by encoding these conditions as an additional channel of the input image; but this is not as simple  
390 for nonlinear damage prediction, unlike linear elastic stress prediction and will require more data  
391 for training and testing.

392 While non-circular carbon fiber reinforced plastics (CFRPs) may have superior performance,  
393 circular CFRPs are most commonly used. Therefore, we only considered circular fibers in 2D  
394 RVEs relying on the assumption that the fibers are well-aligned in the longitudinal (out-of-plane)  
395 direction. The proposed two-stage ML model illustrates that the “image-to-value” CNN model  
396 shows better performance when trained with rasterized damage images compared to the RVE image,  
397 for the same model architecture and hyperparameters. We also varied the model architecture and  
398 hyperparameters (e.g. by considering different initial channels and the number of convolutional  
399 blocks). The corresponding RMSREs given in Table S1 in the supplementary information illustrate  
400 that the two-stage model predictions have smaller RMSRE compared to that from the one-stage  
401 model. This supports our hypothesis that the damage field provides better cues to the CNN model  
402 in predicting the peak load.

403 **CONCLUSION**

404 In this paper, our goal was to formulate and train a computationally inexpensive surrogate for  
405 the phase field damage model, which can quickly predict the damage field and peak load given the  
406 microstructure geometry of a unidirectional fiber-reinforced composite. To this end, we developed  
407 a two-stage CNN-based deep learning model – where the first stage predicts the damage field given  
408 a microstructure image, and the second stage takes in this predicted damage field and predicts the  
409 peak load value. Our preliminary check showed that the model was able to learn the relationship  
410 between fiber distribution, damage field and peak load, with fixed fiber radii and volume fraction.  
411 We next demonstrated that the direct prediction of peak load from the RVE image using the CNN-  
412 based image-to-value model is less accurate compared to the indirect prediction of peak load from  
413 the damage field image. We next assessed the generalization of the surrogate model to predict  
414 damage field and peak load for different fiber radii and volume fractions with different datasets.  
415 Although the model provided reasonably accurate predictions of the damage field and peak load  
416 when training with only the extremes of fiber radii and volume fraction, we find that training with  
417 the range of fiber radii and volume fraction improves model generalization.

418 **DATA AVAILABILITY STATEMENT**

419 Dataset and codes for machine learning model training and testing used during the study are  
420 available in repositories online in accordance with funder data retention policies. (Codes on  
421 GitHub: <https://github.com/YuxiangGao0321/Phase-field-damage-CNN>, Datasets on  
422 Dryad: <https://doi.org/10.5061/dryad.jh9w0vtfp>.) Codes for solving the phase-field  
423 damage problem by FEniCS to generate machine learning datasets of this study are available from  
424 the corresponding author upon reasonable request.

425 **ACKNOWLEDGEMENT**

426 This work was funded by a research grant from 3M Company. YG and RD acknowledge the  
427 financial support from NSF grant #PLR-1847173, which supported the development of the phase  
428 field model and the finite element implementation in FEniCS software.

429 **SUPPLEMENTAL MATERIALS**

430 Figures S1 - S6, Table S1, and twelve datasets are available online [https://doi.org/10.](https://doi.org/10.5061/dryad.jh9w0vtfp)

431 [5061/dryad.jh9w0vtfp.](https://doi.org/10.5061/dryad.jh9w0vtfp)

## REFERENCES

433 Ahmadian, H., Yang, M., Nagarajan, A., and Soghrati, S. (2019). “Effects of shape and misalignment  
 434 of fibers on the failure response of carbon fiber reinforced polymers.” *Computational Mechanics*,  
 435 63(5), 999–1017.

436 Alnæs, M., Blechta, J., Hake, J., Johansson, A., Kehlet, B., Logg, A., Richardson, C., Ring, J.,  
 437 Rognes, M. E., and Wells, G. N. (2015). “The FEniCS Project Version 1.5.” *Archive of Numerical  
 438 Software*, 3(100).

439 Bhaduri, A., Gupta, A., and Graham-Brady, L. (2021). “Stress field prediction in fiber-reinforced  
 440 composite materials using a deep learning approach.” *arXiv:2111.05271 [cond-mat]*.

441 Bharali, R., Goswami, S., Anitescu, C., and Rabczuk, T. (2022). “A robust monolithic solver  
 442 for phase-field fracture integrated with fracture energy based arc-length method and under-  
 443 relaxation.” *Computer Methods in Applied Mechanics and Engineering*, 394, 114927.

444 Croom, B. P., Berkson, M., Mueller, R. K., Presley, M., and Storck, S. (2022). “Deep learning  
 445 prediction of stress fields in additively manufactured metals with intricate defect networks.”  
 446 *Mechanics of Materials*, 165, 104191.

447 Espadas-Escalante, J. J. and Isaksson, P. (2019). “Mesoscale analysis of the transverse cracking  
 448 kinetics in woven composite laminates using a phase-field fracture theory.” *Engineering Fracture  
 449 Mechanics*, 216, 106523.

450 Francfort, G. A. and Marigo, J. J. (1998). “Revisiting brittle fracture as an energy minimization  
 451 problem.” *Journal of the Mechanics and Physics of Solids*, 46(8), 1319–1342.

452 Geuzaine, C. and Remacle, J.-F. (2009). “Gmsh: A 3-D finite element mesh generator with built-in  
 453 pre- and post-processing facilities: THE GMSH PAPER.” *International Journal for Numerical  
 454 Methods in Engineering*, 79(11), 1309–1331.

455 Goswami, S., Anitescu, C., Chakraborty, S., and Rabczuk, T. (2020). “Transfer learning enhanced  
 456 physics informed neural network for phase-field modeling of fracture.” *Theoretical and Applied  
 457 Fracture Mechanics*, 106, 102447.

458 Gupta, A., Krishnan, U. M., Mandal, T. K., Chowdhury, R., and Nguyen, V. P. (2022). “An adaptive

459 mesh refinement algorithm for phase-field fracture models: Application to brittle, cohesive, and  
460 dynamic fracture.” *Computer Methods in Applied Mechanics and Engineering*, 399, 115347.

461 Hirshikesh, H., Pramod, A. L. N., Waisman, H., and Natarajan, S. (2021). “Adaptive phase field  
462 method using novel physics based refinement criteria.” *Computer Methods in Applied Mechanics  
463 and Engineering*, 383, 113874.

464 Ioffe, S. and Szegedy, C. (2015). “Batch Normalization: Accelerating Deep Network Training  
465 by Reducing Internal Covariate Shift.” *Proceedings of the 32nd International Conference on  
466 Machine Learning*, PMLR, 448–456, <<https://proceedings.mlr.press/v37/ioffe15.html>> (June).

467 Khalid, M., Baber, J., Kasi, M. K., Bakhtyar, M., Devi, V., and Sheikh, N. (2020). “Empirical  
468 Evaluation of Activation Functions in Deep Convolution Neural Network for Facial Expression  
469 Recognition.” *2020 43rd International Conference on Telecommunications and Signal Process-  
470 ing (TSP)*, 204–207 (July).

471 Kristensen, P. K. and Martínez-Pañeda, E. (2020). “Phase field fracture modelling using quasi-  
472 Newton methods and a new adaptive step scheme.” *Theoretical and Applied Fracture Mechanics*,  
473 107, 102446.

474 Kuhn, C. and Müller, R. (2016). “A discussion of fracture mechanisms in heterogeneous materials  
475 by means of configurational forces in a phase field fracture model.” *Computer Methods in Applied  
476 Mechanics and Engineering*, 312, 95–116.

477 Lo, Y.-S., Borden, M. J., Ravi-Chandar, K., and Landis, C. M. (2019). “A phase-field model for  
478 fatigue crack growth.” *Journal of the Mechanics and Physics of Solids*, 132, 103684.

479 Miehe, C., Hofacker, M., and Welschinger, F. (2010). “A phase field model for rate-independent  
480 crack propagation: Robust algorithmic implementation based on operator splits.” *Computer  
481 Methods in Applied Mechanics and Engineering*, 199(45), 2765–2778.

482 Mohammadzadeh, S. and Lejeune, E. (2021). “Predicting Mechanically Driven Full-Field Quanti-  
483 ties of Interest with Deep Learning-Based Metamodels.” *arXiv:2108.03995 [physics]*.

484 Pathan, M., Tagarielli, V., Patsias, S., and Baiz-Villafranca, P. (2017). “A new algorithm to generate  
485 representative volume elements of composites with cylindrical or spherical fillers.” *Composites*

486        *Part B: Engineering*, 110, 267–278.

487        Pathan, M. V., Ponnusami, S. A., Pathan, J., Pitisongsawat, R., Erice, B., Petrinic, N., and Tagarielli,  
488        V. L. (2019). “Predictions of the mechanical properties of unidirectional fibre composites by  
489        supervised machine learning.” *Scientific Reports*, 9(1), 13964.

490        Raissi, M., Perdikaris, P., and Karniadakis, G. (2019). “Physics-informed neural networks: A  
491        deep learning framework for solving forward and inverse problems involving nonlinear partial  
492        differential equations.” *Journal of Computational Physics*, 378, 686–707.

493        Ronneberger, O., Fischer, P., and Brox, T. (2015). “U-Net: Convolutional Networks for Biomedical  
494        Image Segmentation.” *Medical Image Computing and Computer-Assisted Intervention (MICCAI*  
495        2015), N. Navab, J. Hornegger, W. M. Wells, and A. F. Frangi, eds., Cham, Springer International  
496        Publishing, 234–241.

497        Sepasdar, R., Karpatne, A., and Shakiba, M. (2022). “A data-driven approach to full-field nonlinear  
498        stress distribution and failure pattern prediction in composites using deep learning.” *Computer  
499        Methods in Applied Mechanics and Engineering*, 397, 115126.

500        Shakiba, M., Brandyberry, D. R., Zacek, S., and Geubelle, P. H. (2019). “Transverse failure  
501        of carbon fiber composites: Analytical sensitivity to the distribution of fiber/matrix interface  
502        properties.” *International Journal for Numerical Methods in Engineering*, 120(5), 650–665.

503        Simonyan, K. and Zisserman, A. (2015). “Very Deep Convolutional Networks for Large-Scale Im-  
504        age Recognition.” *The 3rd International Conference on Learning Representations (ICLR2015)*,  
505        <<http://arxiv.org/abs/1409.1556>> (April).

506        Singh, A. and Pal, S. (2021). “Multi-phase field modeling for various fracture mechanisms in  
507        composites.” *Engineering Fracture Mechanics*, 241, 107348.

508        Sun, X., Duddu, R., and Hirshikesh (2021). “A poro-damage phase field model for hydrofracturing  
509        of glacier crevasses.” *Extreme Mechanics Letters*, 45, 101277.

510        Svolos, L., Berger-Vergiat, L., and Waismann, H. (2020). “Updating strategy of a domain decomposi-  
511        tion preconditioner for parallel solution of dynamic fracture problems.” *Journal of Computational  
512        Physics*, 422, 109746.

513 Svolos, L., Mourad, H. M., Manzini, G., and Garikipati, K. (2022). “A fourth-order phase-field  
514 fracture model: Formulation and numerical solution using a continuous/discontinuous Galerkin  
515 method.” *Journal of the Mechanics and Physics of Solids*, 165, 104910.

516 Tan, W. and Martínez-Pañeda, E. (2021). “Phase field predictions of microscopic fracture and  
517 R-curve behaviour of fibre-reinforced composites.” *Composites Science and Technology*, 202,  
518 108539.

519 Wu, J.-Y. and Nguyen, V. P. (2018). “A length scale insensitive phase-field damage model for brittle  
520 fracture.” *Journal of the Mechanics and Physics of Solids*, 119, 20–42.

521 Yang, Z., Yu, C.-H., and Buehler, M. J. (2021a). “Deep learning model to predict complex stress  
522 and strain fields in hierarchical composites.” *Science Advances*, 7(15), eabd7416.

523 Yang, Z., Yu, C.-H., Guo, K., and Buehler, M. J. (2021b). “End-to-end deep learning method to  
524 predict complete strain and stress tensors for complex hierarchical composite microstructures.”  
525 *Journal of the Mechanics and Physics of Solids*, 154, 104506.

526 Zhang, P., Hu, X., Yang, S., and Yao, W. (2019). “Modelling progressive failure in multi-phase  
527 materials using a phase field method.” *Engineering Fracture Mechanics*, 209, 105–124.

528 Zhou, K., Sun, H., Enos, R., Zhang, D., and Tang, J. (2021). “Harnessing deep learning for physics-  
529 informed prediction of composite strength with microstructural uncertainties.” *Computational  
530 Materials Science*, 197, 110663.

**List of Tables**

1 Details of the various training and test datasets . . . . . 24

2 Error metrics obtained for each test set . . . . . 25

**TABLE 1.** Details of the various training and test datasets

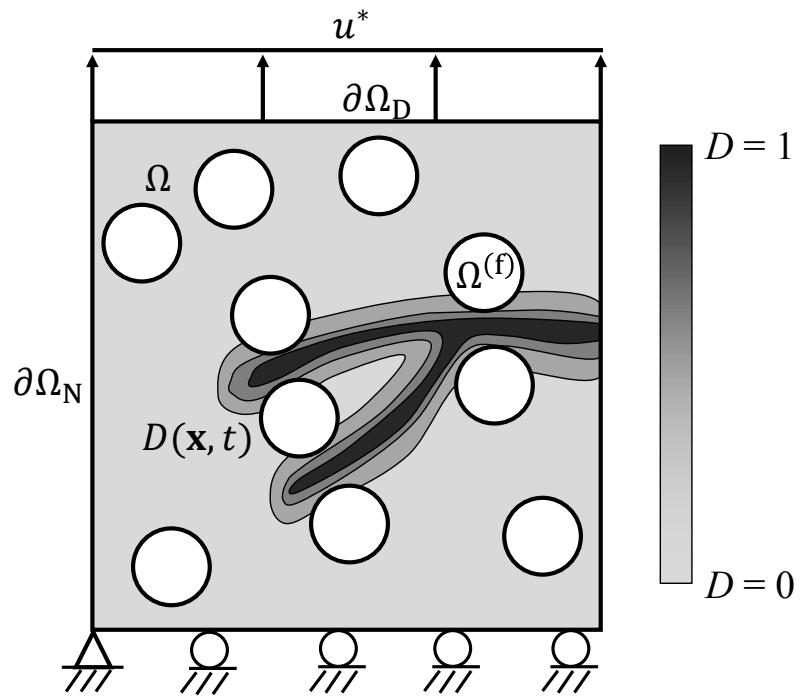
No.	Training set			Test set		
	$V_f$	$R_f(\mu m)$	Size	$V_f$	$R_f(\mu m)$	Size
A	0.5	4	800	0.5	4	200
B	0.5	2, 6	1000+1000	0.5	4	200
C	0.5	2, 4, 6	667+667+667	0.5	4	200
D	0.2, 0.5	4	1000+1000	0.3	4	200
E	0.2, 0.22, ..., 0.5	4	130×16	0.3	4	200

**TABLE 2.** Error metrics obtained for each test set

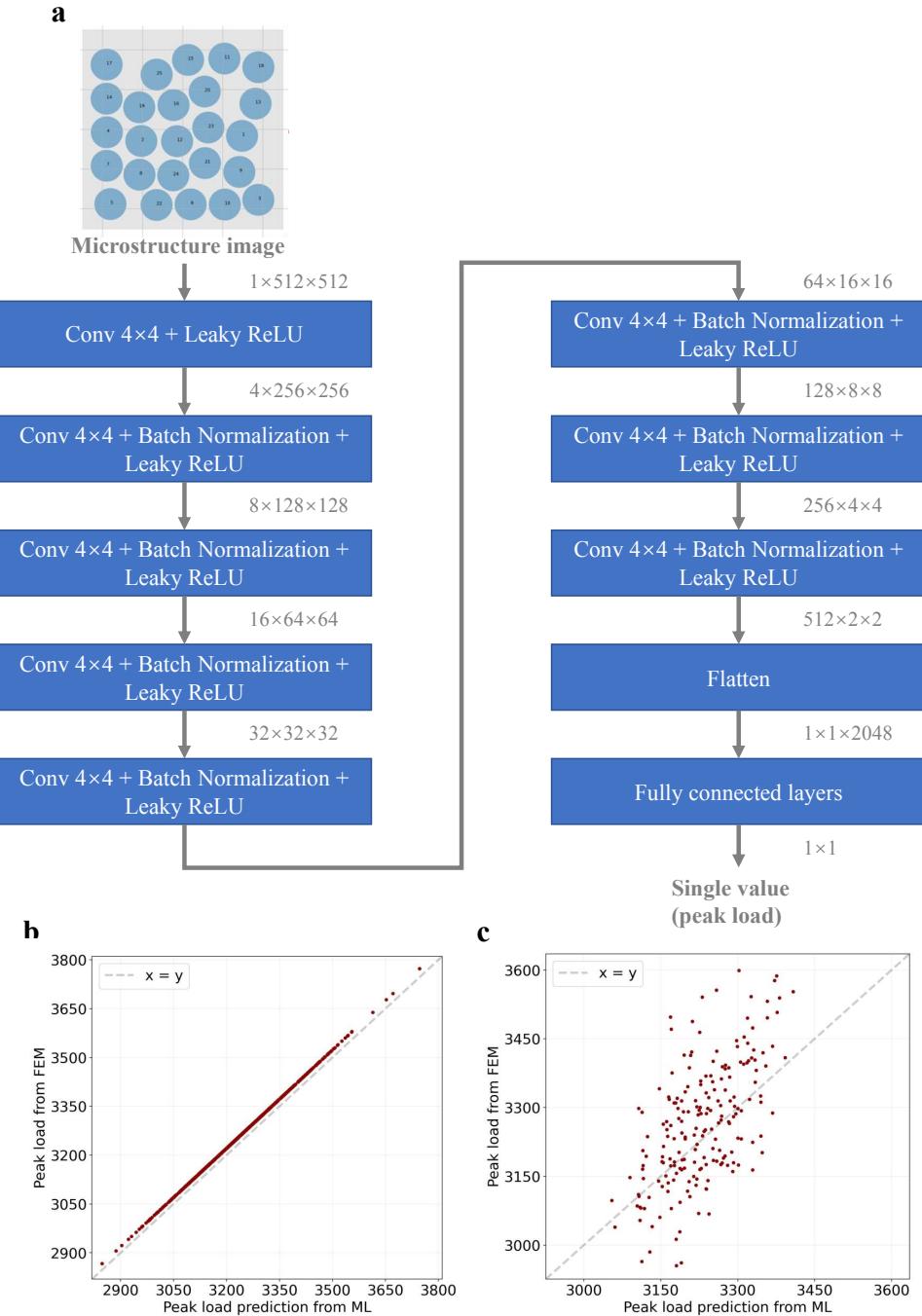
Dataset No.	Avg. NRMSE for damage field prediction	RMSRE for peak load prediction
A	0.1290	0.0212
B	0.1589	0.0262
C	0.1156	0.0172
D	0.1477	0.0327
E	0.1481	0.0267

534 **List of Figures**

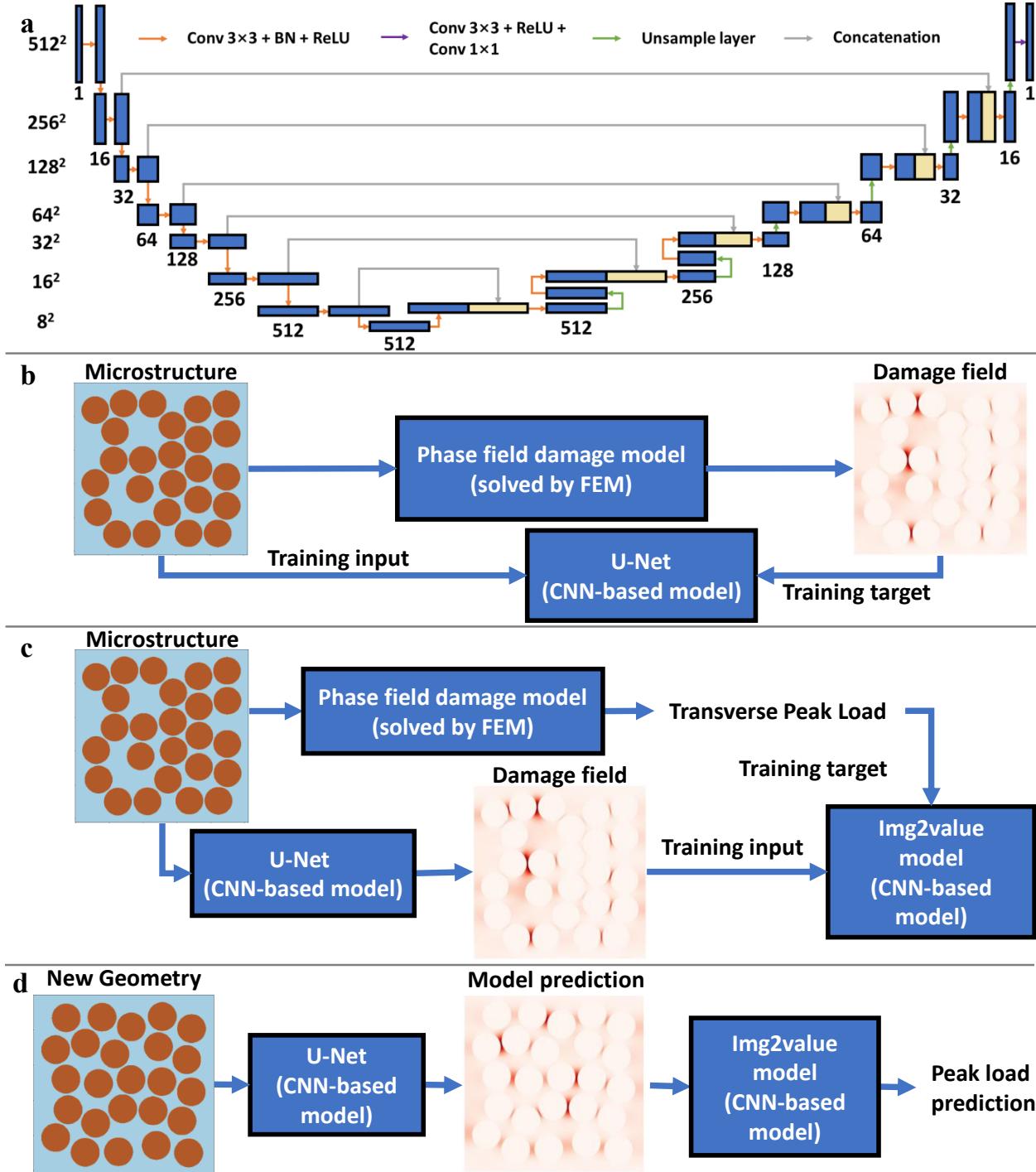
535 1	Schematic of the solid domain $\Omega$ showing the representative volume element (RVE) 536 of a fiber-reinforced composite and a diffused crack interface described by the phase 537 field damage variable $D$ . . . . .	27
538 2	(a) Architecture of the baseline or one-stage CNN-based model conditioned on 539 images of the composite microstructure. Comparison of the peak load predicted 540 by the model, and the peak load calculated by FEM evaluated on: (b) the training 541 set and (c) the test set. . . . .	28
542 3	(a) Architecture of the CNN based model that takes in microstructure geometry as 543 input (left), and predicts its corresponding damage field at peak load (right); (b) 544 training a CNN-based UNet for damage field prediction; (c) training a second CNN 545 model for peak load prediction; (d) Model inference given a new RVE geometry. . . . .	29
546 4	Model performance evaluation on the test set with $R_f = 4 \mu\text{m}$ and $V_f = 0.5$ : Damage 547 field predicted by the UNet model and the ground truth calculated by the FEM, and 548 the difference between them for the case with (a) the smallest NRMSE and (b) the 549 largest NRMSE; (c) The distribution of NRMSE; (d) The distribution of RMSRE 550 for the peak load prediction. The peak load predicted by (e) the surrogate model 551 and (f) the baseline model against that from the FEM. . . . .	30
552 5	The distribution of NRMSE for the damage field predicted by the model trained and 553 evaluated with (a) dataset B; (b) dataset C (evaluated with $R_f = 2 \mu\text{m}$ );(c) dataset 554 C; (d) dataset C (evaluated with $R_f = 6 \mu\text{m}$ );(e) dataset D; (f) dataset E. . . . .	31
555 6	The distribution of relative error for the peak load predicted by the model trained 556 and evaluated with (a) dataset B; (b) dataset C (evaluated with $R_f = 2 \mu\text{m}$ );(c) 557 dataset C; (d) dataset C (evaluated with $R_f = 6 \mu\text{m}$ );(e) dataset D; (f) dataset E. . . . .	32



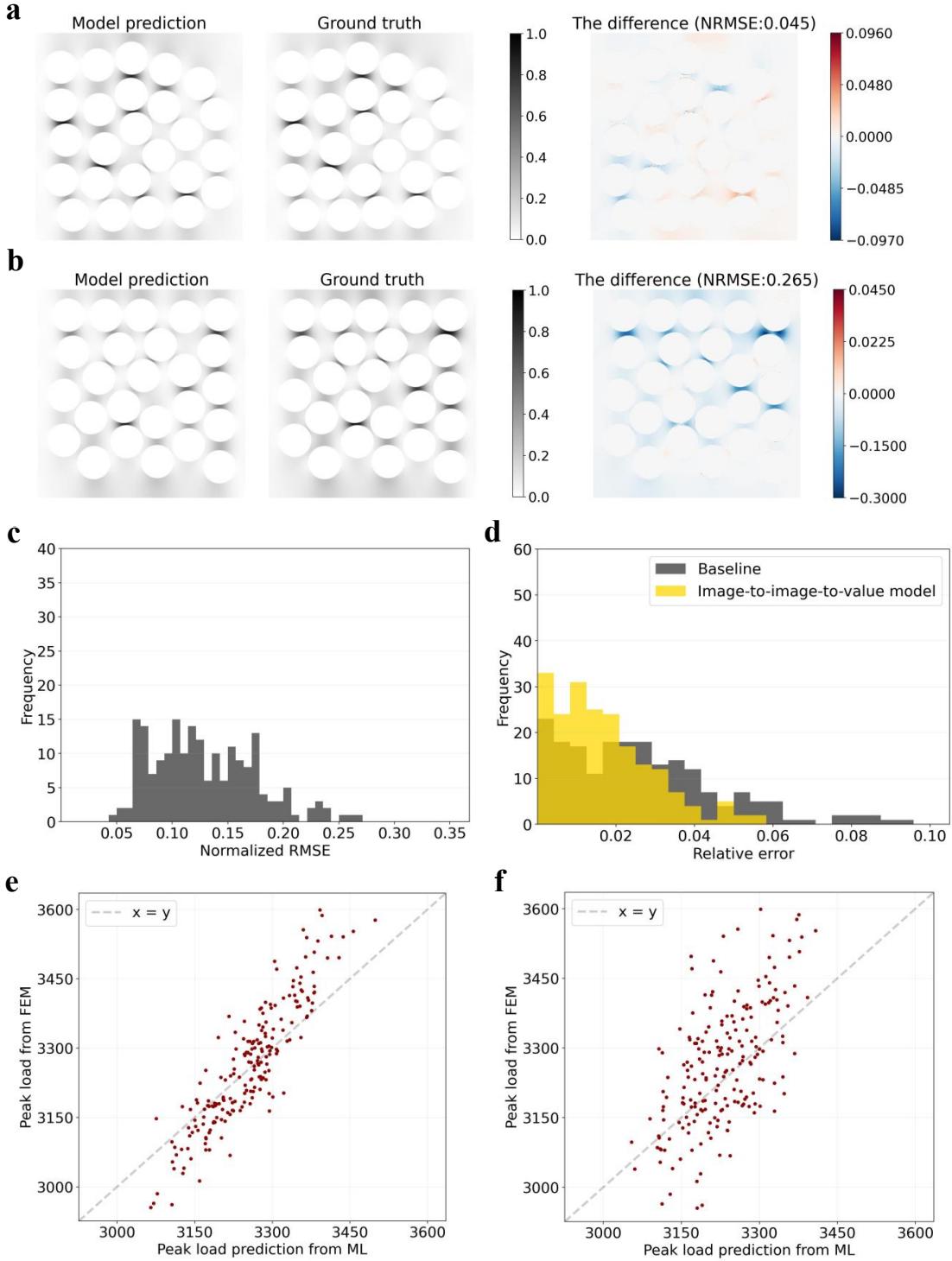
**Fig. 1.** Schematic of the solid domain  $\Omega$  showing the representative volume element (RVE) of a fiber-reinforced composite and a diffused crack interface described by the phase field damage variable  $D$ .



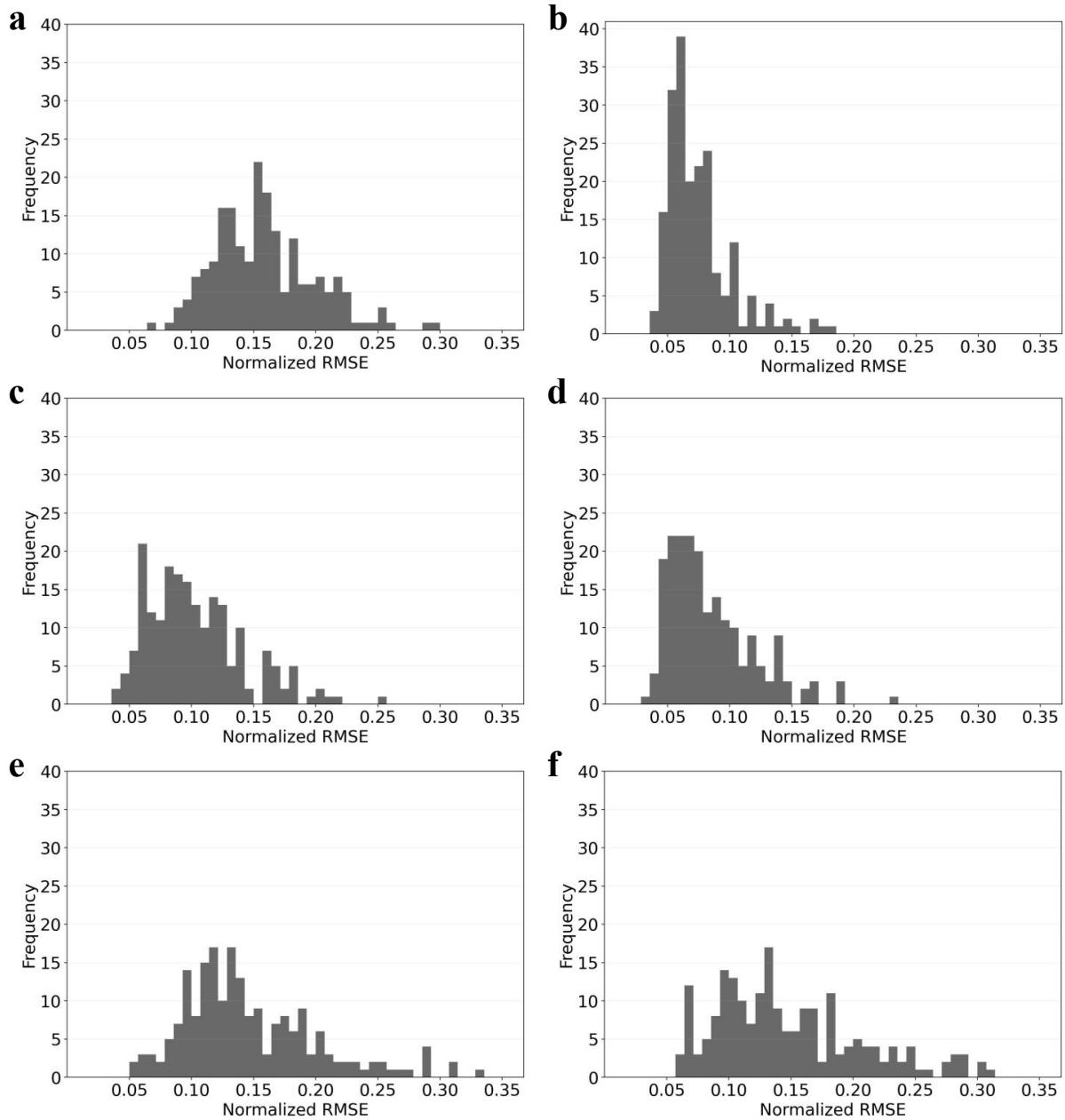
**Fig. 2.** (a) Architecture of the baseline or one-stage CNN-based model conditioned on images of the composite microstructure. Comparison of the peak load predicted by the model, and the peak load calculated by FEM evaluated on: (b) the training set and (c) the test set.



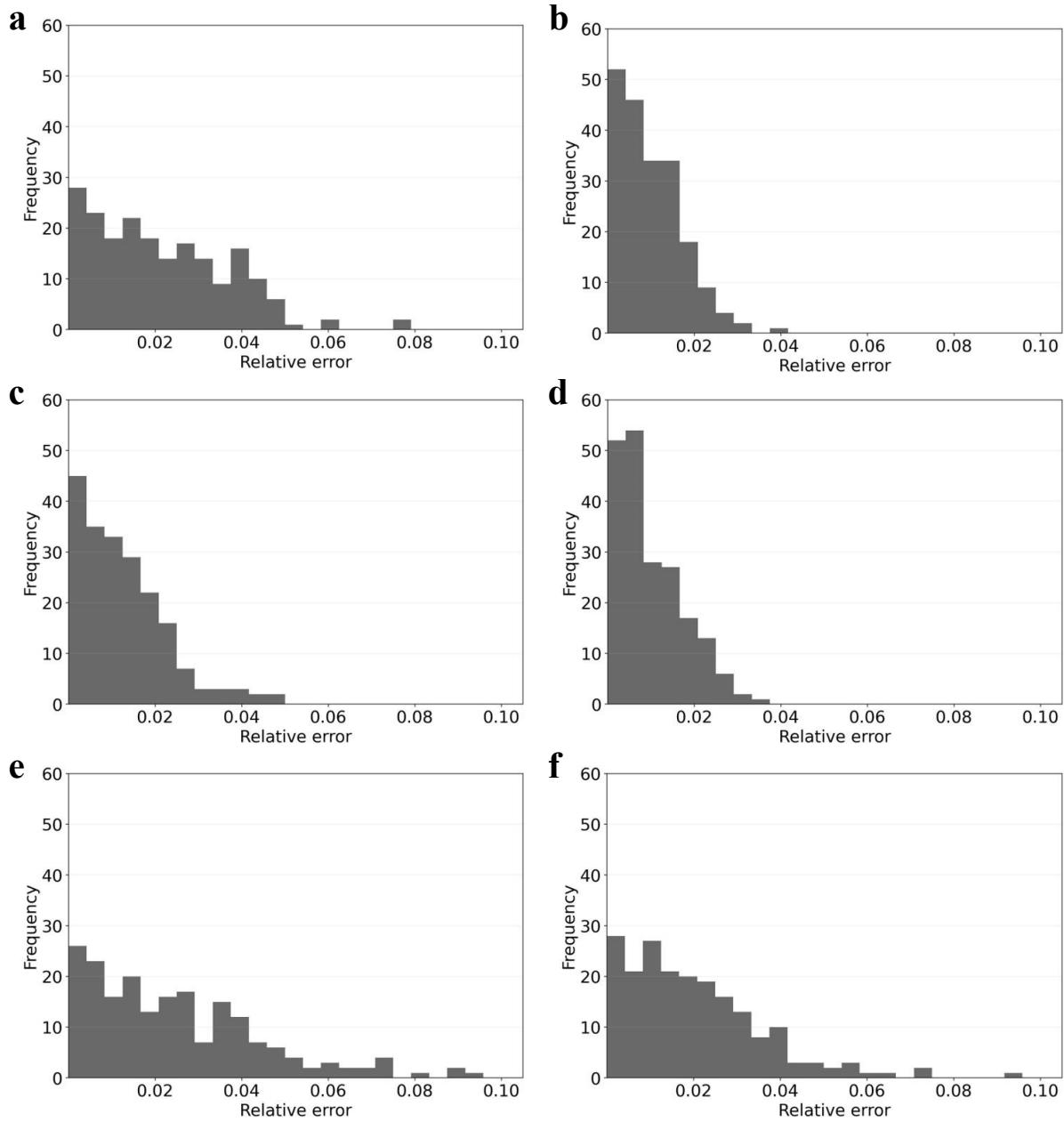
**Fig. 3.** (a) Architecture of the CNN based model that takes in microstructure geometry as input (left), and predicts its corresponding damage field at peak load (right); (b) training a CNN-based UNet for damage field prediction; (c) training a second CNN model for peak load prediction; (d) Model inference given a new RVE geometry.



**Fig. 4.** Model performance evaluation on the test set with  $R_f = 4 \mu\text{m}$  and  $V_f = 0.5$ : Damage field predicted by the UNet model and the ground truth calculated by the FEM, and the difference between them for the case with (a) the smallest NRMSE and (b) the largest NRMSE; (c) The distribution of NRMSE; (d) The distribution of RMSRE for the peak load prediction. The peak load predicted by (e) the surrogate model and (f) the baseline model against that from the FEM.



**Fig. 5.** The distribution of NRMSE for the damage field predicted by the model trained and evaluated with (a) dataset B; (b) dataset C (evaluated with  $R_f = 2 \mu\text{m}$ );(c) dataset C; (d) dataset C (evaluated with  $R_f = 6 \mu\text{m}$ );(e) dataset D; (f) dataset E.



**Fig. 6.** The distribution of relative error for the peak load predicted by the model trained and evaluated with (a) dataset B; (b) dataset C (evaluated with  $R_f = 2 \mu\text{m}$ );(c) dataset C; (d) dataset C (evaluated with  $R_f = 6 \mu\text{m}$ );(e) dataset D; (f) dataset E.