Dilute Limit Alloy Pd-Cu Bimetallic Catalysts Prepared by Simultaneous Strong Electrostatic Adsorption: A Combined Infrared Spectroscopic and Density Functional Theory Investigation

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Abstract

Well-defined single-atom catalytic sites with unique geometric and electronic properties are at the forefront of catalyst research. One type of SAC is a so-called dilute limit alloy, where single metal atom sites are supported on (or in) the surface of a second metal. This contribution explores the simple and scalable approach of strong electrostatic adsorption (SEA) to synthesize silica-supported dilute limit alloy (DLA) of palladium on copper. The catalysts were characterized using X-ray diffraction, temperature programmed reduction and TEM images to confirm the presence of tightly distributed metal nanoparticles. The isolated Pd sites on Cu were probed using transmission Fourier-transform infrared (FTIR) spectroscopy during adsorption of CO. Through spectral curve-fitting to extract Gaussian peak parameters, coupled with density functional theory (DFT) calculations, the nature and structure of the Pd sites was determined. In addition to typical vibrational features observed for CO on copper, three additional peaks at 2046, 2021 and 1908 cm^{-1} were observed for the dilute limit alloy Pd-Cu system. The DFT analysis of the same system showed that these peaks are consistent with CO linearly adsorbed on an embedded palladium atom on a Cu(100) surface, a Pd on a Cu(111) surface, and a bridge CO between Pd and Cu with the Pd embedded on a Cu(211) surface, respectively. Experimental and computation trends in CO vibration wavenumber could be well interpreted using the Dewar-Chatt-Duncanson (DCD) model. The promise of the SEA approach for producing DLAs and the utility of this combined FTIR/DFT approach for their characterization will be discussed in light of these findings.

1. Introduction

The promise of atom-efficient utilization of precious metals combined with the potential for unique catalytic applications drives recent efforts to develop novel synthesis methods and characterization techniques for single-sites heterogeneous catalysts (SSHC).[1,2,3–5] Such well-defined (and more uniform) active sites may also allow for a more straightforward interpretation of reaction kinetics and mechanisms through experimental evaluation and computational methods.[6,7] Isolating a known active metal (e.g., Pd, Pt, etc.) in a "sea" of another metal is a known strategy to produce a type of single-site catalyst. Such DLA might have unique properties if the electronic structure of the isolated metal is altered or the resulting isolated site geometry causes an ensemble effect. Decreased utilization of the active (often precious) metal with similar or enhanced catalytic performance would increase atom efficiency.

Several reports have appeared in the literature that show the effectiveness of DLA catalysts for a range of reactions. For example, pioneering work by Flytzani-Stephanopoulos and co-workers explored small surface concentrations of palladium atoms dispersed on a Cu(111) surface and alumina-supported Cu particles ($Pd_{0.18}Cu_{15}/Al_2O_3$). Significant improvement was observed for the selective hydrogenation of phenylacetylene to styrene relative to monometallic palladium catalyst.[8] The higher selectivity was attributed to spillover of Pd site-dissociated hydrogen atoms to the Cu surface, thus allowing the inactive copper to serve as an effective bifunctional hydrogenation catalyst. Silica-supported Pd-Cu DLA prepared in the same manner were also found to be active in selective oxidation of methanol towards methyl formate at low temperatures. In this case, the high selectivity is due to the formation of an atomically dispersed Pd oxide active site on the copper matrix.[9] Selective oxidation of methacrolein with methanol to methyl methacrylate was also enhanced by doping isolated Ni atoms on the Au surface. The single atom of nickel acted as

3

59 a binding site for methanol, as a base to stabilize the reaction intermediate and increase the number

60 of Au-Ox active sites.[10] Improvements in dehydrogenation (NiCu), selective partial hydrogenation

61 (PtCu), selective hydrogenation (PdAg, PdAu), and hydrogenolysis (PtCu) were also made with

62 the use of DLA catalysts.[11–15]

63 These promising results using DLAs have motivated researchers in both computational and

64 experimental catalysis.[16,17] Computational studies readily allow for exploring metal combinations

65 to target a specific reaction and can also aid in the interpretation of experimental results.[18,19]

66 However, while the structures of single metal alloy active sites are relatively simple (compared to

67 standard bimetallic catalysts), the synthesis of such structures is challenging. A variety of methods

68 have been used with varying success and complexity. Galvanic displacement (GD) has been

69 successfully applied to make a variety of single metal alloy catalysts.[8,12,20] For example, Flytzani-

70 Stephanopoulos et al. dispersed Pd atoms alloyed on a Cu metal nanoparticle surface on alumina

71 support via galvanic displacement of Cu by Pd.[8] The method was extended to prepare NiCu[8] and

72 PtCu[9,10] DLA nanoparticles supported on silica and alumina, respectively. One limitation of this

73 method is the reduction potential compatibility of the two metals involved. Second, the metal to

74 be displaced must be present in a reduced state, requiring inert gas protection during the GD

75 process.

76 Another method has been proposed by Zhang and co-workers (2015), which involves the

77 formation of hydroxide species $M(OH)_z$ on a reducible oxide support A_xO_y. Calcination then forms

78 M-O-A bonds between the singly dispersed $M(OH)_z$ and A_xO_y, followed by a controlled reduction

79 to remove some oxygen atoms to form isolated M_1A_n sites.[22,23] The reduction of precursors is

80 crucial in preparing the catalyst, and the approach is limited to reducible oxide supports. In the

81 vapor phase, atomic layer deposition of a second metal onto a metal nanoparticle has been

4

82 explored.[24–27] For example, Wang et al. used a combination of deposition-precipitation of Ni

83 followed by selective atomic layer deposition of Pd to produce single metal alloy PdNi on silica.

84 The ALD method was able to selectively target the Pd onto the Ni surface rather than on the silica

85 support surface. However, this selective ALD procedure has only been demonstrated for silica

86 support.[28]

87 Recently, Filie et al. (2021) prepared a dilute PdAu nanoparticles supported on raspberry-

88 colloid-templetated (RCT) silica. It is a three-step synthesis method involving intitial synthesis of

89 monometallic and narrowly-sized (~5nm) gold nanoparticles (NPs). Dilute bimetallic Pd_xAu_{1-x}

90 was prepared by adding palladium (II) nitrate hydrate $(Pd(NO_3)_2)$ to a solution containing as-

91 synthesized Au NPs and ascorbic acid aqueous solution. Lastly, these bimetallic nanoparticles

92 were then delivered to silica support using the RCT approach. [29]

93 More traditional co- and sequential- incipient wetness (IW) impregnation methods have also

94 been used to make single metal alloy catalysts.[30–35] For example, Pd-Ag/SiO_2 with dilute levels of

95 Pd were synthesized using nitrates as precursors.[36] The resulting improvement in activity and

96 selectivity for hydrogenation of acrolein was attributed to the addition of atomically dispersed Pd.

97 However, as is typical with IW methods, the particle sizes were on the order of 5-9 nm, which is

98 not a high degree of dispersion.

99 One key consideration in designing a single atom alloy catalyst is the possible segregation of the

100 active isolated surface atom into the bulk phase of the abundant host metal. Zhang et al. (2015)

101 reported surface atom migration during the galvanic replacement of Pd on Ag metal surface due

102 to the large difference in surface free energy. Pd tends to diffuse to the bulk of the base Ag particle

103 and vice-versa, providing fresh Ag atoms for galvanic displacement.[13] Thermodynamic limitations

104 depending on the surface free energies dictate how a given metal would preferentially segregate

105   to a nanoparticle surface, thus possibly avoiding bulk migration. In addition, for a given reaction,

106   the strong interaction of a single metal site with a specific adsorbate (e.g., CO) might serve to

107   stabilize the atom on the surface. In the context of ultrasmall bimetallic catalysts as prepared by

108   Wong et.al. (2017), the significantly low number of bulk atoms provides a low probability for an

109   isolated surface atom to bulk segregate. In a cuboctahedron cluster model developed by Benfield

110   (1992), a copper particle having a 1.49 nm particle diameter (m=4, a=0.362 nm) has ~63% of its

111   atoms on the surface.[37] This high probability of surface atoms as the particle size decreases is the

112   motivation for the synthesis of isolated surface atoms using the co-SEA method. This strategy

113   would allow for facile one-pot synthesis of any dilute limit alloy bimetallic catalyst (DLA) using

114   the method of simultaneous strong electrostatic adsorption (co-SEA).[38]

115   One challenging aspect of isolated metal atoms is characterization due to their inherent small

116   concentration in the catalysts. X-ray photoelectron spectroscopy (XPS) typically does not provide

117   a high enough signal for the isolated atoms and would require a synchrotron beam line for a

118   brighter X-ray source.[39] Aberration corrected scanning transmission electron microscopy (AC-

119   STEM) can be used to distinguish between a single atom and neighboring atoms when there is a

120   high atomic number difference (i.e., Z-contrast) but is often challenging when the atoms are similar

121   in size.[40] Other techniques such as temperature programmed reduction (TPR), may provide

122   information about the interaction between the dilute metal and its host, but not the direct surface

123   environment.[41,42]

124   One approach that shows some promise for characterizing single atom alloy catalysts is Fourier

125   transform infrared spectroscopy (FTIR) of CO adsorption. The highly sensitive signal can allow

126   the vibrational properties of various single atom sites to be explored. FTIR is a well-established

127   method used to characterize monometallic and bimetallic catalyst surfaces. A significant number

128   of publications have been published for most of the metals used in catalysis. Sheppard and Nguyen

129   (1978) consolidated IR results from various studies (single crystals and oxide supported metals)

130   and suggested schemes for interpreting spectral results. In the recent review made by Sheppard

131   and De la Cruz on the reliability of vibrational spectroscopy for structure identification of

132   chemisorbed species such as CO on (111) surfaces, they concluded a 2130-2000 cm^{-1} range for

133   linear (on-top) sites and an extended range 1950-1800 cm^{-1} for 3-fold (hollow) bridge CO

134   overlapping with 2-fold bridge CO in the range 2000-1870 cm^{-1} at higher coverage.[43] FTIR

135   characterization of CO adsorption is relatively straightforward for monometallic catalysts since

136   the CO surface species are associated with only one metal (albeit with various coordination and

137   geometric structures). For a bimetallic catalyst with only one metal adsorbing CO, spectra analysis

138   can be similar to that of the monometallic counterpart[13]. However, in systems where both metals

139   adsorb CO, the resulting spectra involve a convolution of vibrational features that can make even

140   a qualitative analysis challenging.[44,45]

141   In this study, silica-supported Pd-Cu bimetallic catalysts were synthesized using a dilute limit

142   approach involving simultaneous strong electrostatic adsorption (co-SEA).[41,42] The molar ratio of

143   Pd to Cu was varied in an attempt to achieve single atom alloy sites of Pd on Cu. These catalysts

144   have been characterized by a range of structural approaches, and especially with FTIR during CO

145   adsorption, to probe the evolution of the Pd_1Cu single metal alloy sites. Extensive spectral curve

146   fitting over a range of conditions was used to identify unique vibrational features, which were then

147   confirmed using density functional theory (DFT) modeling of various adsorbed species. This

148   powerful combination of experimental and computational approaches revealed the presence of two

149   types of single atom Pd_1Cu species that appear in the dilute limit.

150   **2. Experimental and Computational Details**

2.1 Experiments.

Catalyst Preparation. Evonik Aerosil 300® (BET=304 m^2/gm) was used as silica support for adsorption with a surface loading of 1000 m^2/L. Metal precursors used are copper (II) nitrate hydrate (Sigma-Aldrich 99.999%) and tetraamminepalladium (II) chloride monohydrate (Sigma-Aldrich 99.99%). Dilute limit alloys of palladium-copper supported on silica were prepared by simultaneous strong electrostatic adsorption (co-SEA), which has been described previously.[41] Silica supported monometallic palladium, and copper catalysts were also prepared using the same method but using only the associated single metal precursors. Silica was poured into an ammonium hydroxide-adjusted solution (pH 12) containing cationic palladium and copper ammine complex precursors. In making the dilute limit alloy, a palladium to copper metal ratio range from 1:15 to 1:60 with an initial metal concentration based on one monolayer theoretical surface adsorption density ($\sim$1.2 μmole/m^2) for metal ammine complexes were prepared.[42,46] Silica having a PZC=3.6 deprotonates at pH > PZC and develops the negatively charged surface, thereby allowing for the strong adsorption of one monolayer of hydrated cationic ammine complex on the surface. The mixture was placed in an orbital shaker for 15 minutes, and the support was recovered using vacuum filtration. The recovered silica containing the adsorbed precursors was dried at room temperature for at least 12 hours, followed by a muffle furnace oven drying at 120 °C for 4 hours. Finally, the catalysts were reduced in a horizontal tubular furnace under a gas mixture of 20% H_2 balanced N_2 with a total flow rate of 500 SCCM. A ramp of 5 °C/min to 400 °C and a soak time of 1 hour were employed.

Catalyst Characterization: Aliquots of solution were taken before and after contact with the support to determine metal adsorption. Metal concentrations were measured using a PerkinElmer Avio 200 inductively-coupled plasma optical emission spectrometer (ICP-OES) using

174   multiwavelength calibration, multiple replicate data, and multiple view mode. The samples were

175   diluted to minimize the effect of highly volatile compounds on plasma stability.

176   X-ray diffraction (XRD) patterns were obtained using a Rigaku MiniFlex II equipped with a

177   high sensitivity D/tex Ultra Si slit detector. Patterns were recorded from 10-80° 2θ using a step

178   size of 0.02°, and a Cu-Kα radiation source (λ=1.5406 Å) operated at 30 mA and 15 kV.

179   X-ray photoelectron spectroscopy (XPS) investigation of untreated and treated catalysts was

180   carried out using a Kratos AXIS Ultra DLD XPS (Kratos Analytical). The XPS system is equipped

181   with a monochromatic Al Kα source operated at 15 keV and 150 W, a hemispherical analyzer,

182   charge neutralizer, catalyst treatment cell, and a load lock chamber for rapid introduction of

183   samples without breaking vacuum. The X-rays were incident at an angle of 45°, with respect to

184   the surface normal. Analysis was performed at a pressure of ~1 × 10^{-9} mbar and high-resolution

185   core level spectra were measured with a pass energy of 40 eV. The XPS experiments were

186   performed by using an electron beam, directed on the sample, for charge neutralization. The curve

187   fitting procedure was carried out using the XPS Peak 41 software and the peak approximation was

188   carried out by a combination of Gaussian - Lorentzian functions, with subtraction of Shirley-type

189   background. The in-situ reduction of the catalysts was performed in a reaction cell (Model: ES-

190   009R01) directly attached to the XPS chamber, which allows the sample to be treated at gas flow

191   conditions. The samples were transferred inside the reaction cell and back to the analysis chamber

192   without exposure to the atmosphere.

193   Scanning transmission electron microscopy (STEM) images were obtained with a JEOL 2100F

194   200kV FEG-STEM/TEM equipped with a CEOS Cs corrector on the illumination system. The

195   equipment was controlled to have an aperture of 17.5 mrad, which at 200 kV provides a nominal

196   probe size of<0.1 nm. High angle annular dark-field (HAADF) STEM images were acquired on a

197   Fischione Model 3000 HAADF detector with a camera length such that the detector spanned 50–

198   191 mrad. The dwell time per pixel was 15.8 μs. The samples were prepared for STEM by dropping

199   the powder onto 300 mesh Cu grids coated with a holey amorphous carbon film.

200   TPR was performed on the oven-dried catalysts by using a Micromeritics 2920 equipped with

201   a TCD. The temperature was set at a ramp rate of 5°C/min, and the reducing gas of 10% H_2

202   balanced Ar was set at a flow rate of 50 SCCM.

203   A Nicolet Nexus 470 spectrometer equipped with liquid nitrogen-cooled MCT-B detector was

204   used in transmission to obtain a series of transient CO adsorption spectra with a resolution of 2

205   cm^{-1} averaging 32 scans using a single beam mode. Catalyst samples were compressed to obtain

206   self-supporting 12 mm thin disks with a density for a given thickness of approximately 25 mg/cm^2.

207   The sample is contained inside a transmission cell made of 10 cm long stainless steel capped with

208   IR-transparent NaCl windows on both ends that are cooled with flowing coolant. The cell body is

209   wrapped with heating tape and is heated using a temperature controller. A thermocouple is located

210   near the sample to monitor temperature during sample pretreatment and measurement. In situ

211   drying and reduction pretreatment were performed prior to each measurement. Samples were dried

212   for 1 hour at 100°C with flowing nitrogen gas. Upon cooling to room temperature, the samples

213   were then reduced at 400°C for 1 hour in a 20% H_2 balanced N_2 gas mixture with a total flow rate

214   of 250 SCCM. A ramp rate of 5°C/minute was used in both the drying and reduction steps and

215   only the cooling steps use nitrogen gas. A series of spectral measurements, with background

216   spectra measured with nitrogen gas flowing, were recorded in 3 minutes intervals starting with the

217   introduction of 1% CO/He. The first stage (CO adsorption) involves measurements starting with

218   the introduction of CO gas into the system until saturation condition within 90-120 minutes. The

219   second stage (N_2 purging) starts upon replacing the CO gas mixture with nitrogen gas until a final

220　purge spectrum at approximately 90-120 minutes. All measurements were made under

221　atmospheric pressure and room temperature.

222　　Spectral fitting using Gaussian functions was made according to scheme 1. Prior to fitting the

223　dilute limit alloy spectra, the corresponding monometallic spectra were deconvoluted and fitted

224　first. A review of the adsorbed CO vibration wavenumber found in the literature was made for

225　each metal to determine the spectral similarities and peaks observed. Curve fitting for the series of

226　spectra was made using a general purpose curve fitting program Fityk[47]. The optimized Gaussian

227　function widths (FWHM) of the relevant peaks are kept the same across a given series of spectra.

228　The peak center wavenumber and height/area were allowed to change from spectrum to spectrum.

229　Thus, for the purpose of analysis, it was assumed that the FWHM values of the contributing

230　adsorbed CO peaks do not change as a function of their intensity. In the case of spectra from the

231　Pd-Cu system, the obtained Gaussian functions from the monometallic copper fitting were used as

232　part of the starting set of fitting functions for the bimetallic catalyst spectra. The same FWHM and

233　peak ratios from monometallic copper were used to obtain the contributions from adsorbed CO on

234　copper. Additional Gaussian functions associated with isolated palladium were then included in

235　the fitting using the series of spectra obtained during the N_2 purging stage. The set of Gaussian

236　functions for monometallic palladium were also included in the fitting function to determine any

237　presence of isolated monometallic palladium nanoparticles. All deconvoluted spectra presented on

238　this paper converge with an R-square above 0.99 and a residual standard deviation of less than

239　0.002.

240　Scheme 1 General fitting scheme for dilute limit alloy bimetallic catalyst.

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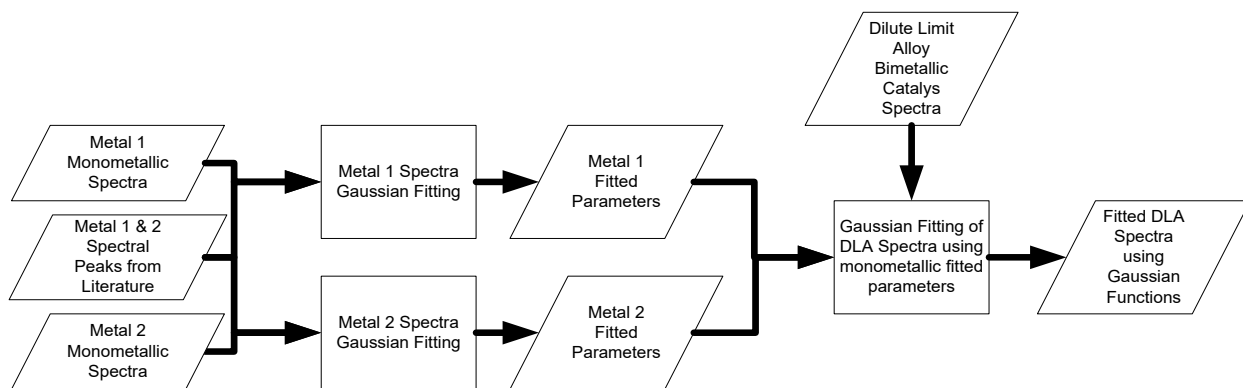
2.2 Computational Methods

First principle calculations were performed using non-spin polarized plane-wave Density Functional Theory (DFT) as implemented in the Vienna Ab-initio Simulation Package (VASP 5.4.4).[48,49] A frozen-core, all-electron projector augmented wave (PAW)[49,50] approach was used to describe the electron-ion interactions. The Revised Perdew-Burke-Ernzerhof (RPBE) functional proposed by Hammer *et al*.[51] was used to treat the exchange-correlation effects. Compared to other functionals within the Generalized Gradient Approximation (GGA), RPBE functional is reported in the literature to predict accurate adsorption geometries for adsorbed CO over transition metals.[52–54] In particular, RPBE predicted experimentally observed CO adsorption sites over several facets of Cu.[55]

Basis sets included wavefunctions with kinetic energy up to 600 eV. A 1.0×10^{-7} eV convergence criterion was set for the electronic Self-Consistent Field (SCF) loops. Structures were considered relaxed when the maximum force on any atom was less than 0.01 eV/Å. Harris corrections based on the Harris-Foulkes formalism[56,57] have been applied to the forces and stress tensors, and the total energy was corrected for dipole effects using a modified version of the Makov-Payne scheme.[58] The Brillouin zone was sampled using a Monkhorst-Pack[59] $4 \times 4 \times 1$ k-point grid. First-order Methfessel-Paxton[60] smearing ($\sigma = 0.10$ eV) was used to speed up the

260   convergence of reciprocal space integrals with respect to the number of k-points. Bulk lattice

261   constants for FCC-Cu and FCC-Pd were calculated to be (a_{Cu} = 3.672 Å) and (a_{Pd} = 3.972 Å),

262   values that are in good agreement with reported experimental values of (a_{Cu} = 3.597 Å) and (a_{Pd}

263   = 3.859 Å).[61] Different CO-metal surface complexes were modeled using slab models consisting

264   of 4 surface layers. For the (111) and (100) surface models we used supercell of size (3 × 4) . A

265   (2 × 3) supercell was used for the (211) and (110) surface models. For the embedded site models,

266   a single Cu surface atom was replaced with a single Pd atom. For ad-atom site models a single Pd

267   atom was located over the surface in the energetically most favorable position. For all surface

268   models, the Pd surface concentration is 1/12 ML (1/13 ML for the ad-atom models), except for

269   (211) where we have a surface concentration of 1/18 ML. The top two layers were relaxed in all

270   calculations while the bottom two layers were fixed to their bulk positions. A vacuum space of 15

271   Å was added along the perpendicular to the surface normal to ensure that the charge density trails

272   off to zero along the surface normal. Such a vacuum gap is also sufficient to ensure that slabs do

273   not interact with their periodic images in the vertical direction.

274   Harmonic vibrational frequencies were calculated using a numerical computation of the Hessian

275   matrix using the tools implemented in the VTST package for VASP developed by the Henkelman

276   group.[62] Numerical calculations needed to compute second derivatives of the energy were

277   performed using the central-difference approximation, wherein the adsorbate atoms were displaced

278   by 0.005 Å from their equilibrium positions. To account for the systematic deviation between

279   DFT-calculated vibrational frequencies and experimental frequencies[63,64], a scaling factor of

280   1.0314 was applied to the RPBE frequencies. The scaling factor was selected such that the scaled

281   calculated C—O stretching frequency in the gas phase is equal to the experimental value of 2170

282   cm^{-1}.[65]

3. Results and Discussion

3.1 Characterization of Catalysts

Calculated nominal weight loadings and surface densities of as-synthesized monometallic palladium and copper catalysts and dilute limit alloy Pd-Cu bimetallic catalysts are shown in Table 1. The weight loading of monometallic palladium is comparable with that of the bimetallic catalysts. Applying the synthesis method of Wong et al. (2017), the adsorption of the copper complex on an amorphous silica surface falls within the theoretical monolayer predicted for strong electrostatic adsorption of ammine complex precursor (~1.2 μmole/m^2).[46] This suggests the absence of copper dimer formation as observed in previous works wherein the amount of adsorbed copper is beyond one monolayer of theoretical adsorbed precursor retaining one layer of hydration sheath.[42,66] The observed total surface density also conforms with the maximum density observed in the uptake survey for the Pd-Cu co-SEA system.[41]

Table 1. As synthesized silica-supported Pd, Cu and dilute limit Pd-Cu catalyst weight loading

Catalyst	Measured Molar Cu/Pd Ratio	Weight Loading, wt%		Surface μmole/m^2		Density,
		Pd	Cu	Pd	Cu	Total
Pd		0.09		0.028		0.028
Cu			2.38		1.217	1.217
1Pd-15Cu	19/1	0.19	2.21	0.06	1.13	1.187
1Pd-30Cu	36/1	0.11	2.35	0.034	1.205	1.238
1Pd-60Cu	74/1	0.05	2.34	0.016	1.197	1.213

Within the limit of high-resolution XRD, the powder diffraction pattern (Figure 1) for both the SEA-prepared monometallic and bimetallic catalysts show the absence of sharp peaks for the

14

299   palladium and copper grain. In comparison, a similar loading monometallic copper prepared by

300   dry-impregnation shows the presence of a larger copper grain size (Figure 1). The pattern suggests

301   a small particle size (<2 nm) typical for catalysts prepared by SEA due to the strong metal

302   precursor-support interaction that limits atom migration during nanoparticle formation.[41,67] The

303   particle size also conforms with the STEM image obtained for the bimetallic Pd-Cu system (Figure

304   S1).

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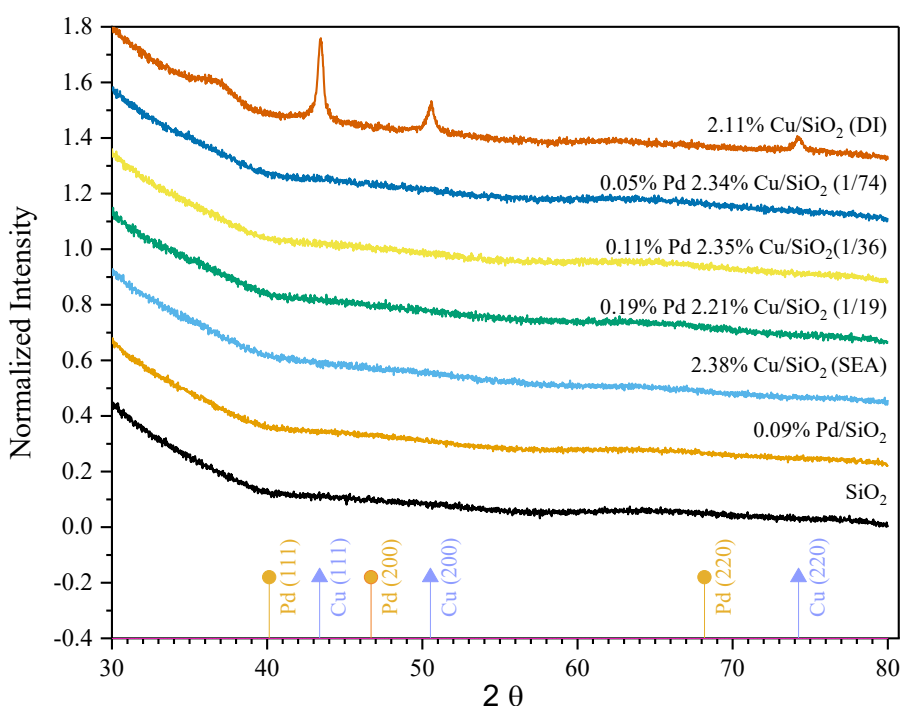
307   Figure 1. Powder X-Ray Diffraction Pattern for Cu, Pd and Dilute Limit Alloy Pd-Cu Catalysts.

308   The H_2-TPR profiles provide insight into how the strongly adsorbed precursors become reduced

309   to form metal particles. In Figure 2, the peak reduction temperature of a series of increasing

310   dilution of palladium in copper shows that the reduction peak position shifts towards that of

311   monometallic copper. Nevertheless, at a dilute amount of palladium, the reduction peak of the

312   copper metal precursor is decreased significantly (~15 °C), confirming a close interaction between

313　the two metals. Since the palladium precursor reduces first (peak at 156 °C), these reduced Pd

314　atoms dissociate H_2 to H atoms[68], which spillover and reduce the copper at a lower temperature.
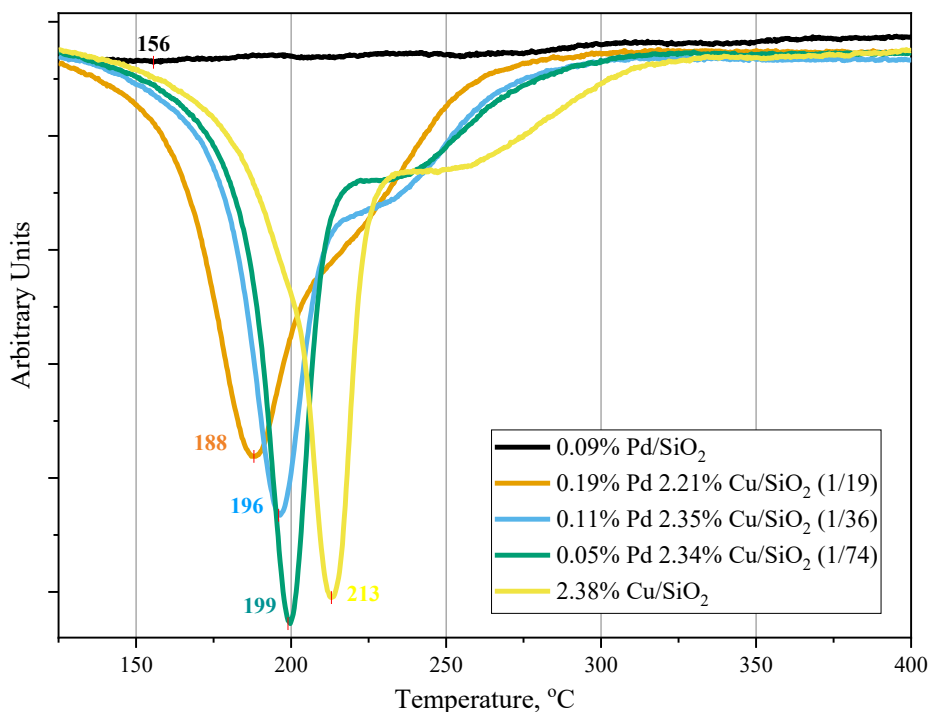

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316　Figure 2. H_2-TPR of Pd, Cu and dilute limit alloy Pd-Cu dried catalyst prepared by strong

317　electrostatic adsorption.

318　　XPS measurements show identical behavior in Cu 2p binding energy for both the monometallic

319　copper and bimetallic dilute limit alloy catalyst. Two peaks were observed for $Cu(OH)_2$ (B.E. =

320　936.4 eV)and CuO (B.E. = 934.1 eV) species for pre-reduced samples exposed to atmospheric

321　conditions. After in-situ reduction, only one peak corresponding to a fully reduced copper (Cu^0)

322　(BE=932.5 eV) was observed. XPS measurement of this reduced copper after CO exposure shows

323　two peaks with higher binding energy (Figure S2a & S2c). The 933.1 eV peak is associated with

324　bridge CO while the 934.6 eV peak is attributed to linear CO. The adsorbed CO on the copper

325　surface has its anti-bonding π bond orbitals filled by electron π-backdonation, thereby losing

326 electrons from the copper metal surface. In an ambient pressure XPS study by Zhang and Ptasinska

327 (2016) on Cu crystals, the shift in Cu 2p was not observed but instead the existence of shake-up

328 peaks suggests CO interaction with Cu.[69] The same increase is observed for the Pd 3d binding

329 energy after CO exposure in the present study (Figure S2b & S2d). An increase in B.E. of Pd 3d

330 from 335.2 eV for the reduced isolated Pd on copper to B.E. 335.9 eV with the Pd-CO system. In

331 a NAP (near ambient pressure)-XPS study by Bukhtiyarov et al. (2018) on Pd-Au bimetallic

332 catalysts, they found the same blue shift in B.E. from Pd state (B.E. =334.7 eV) to Pd-CO state

333 with linear and bridge CO contribution (B.E=335.6 eV and B.E.= 336.2 eV, respectively).[70,71] The

334 Pd-Cu DLA also shows the same trend as that of monometallic Cu, which suggests the absence of

335 electron transfer between Pd and Cu atoms. This could be attributed to strong metal-support

336 interaction that overshadows the electron transfer between Pd and Cu atoms. Spectra for C 1s

337 suggest an adsorbed CO giving a binding energy at ~289.8 eV (Figure S3). The increase in oxygen

338 atomic % in all samples upon CO exposure comes from the CO adsorbed on the metal surface

339 (Table S1-S3). The increased C-C peak after CO exposure for both monometallic copper and Pd-

340 Cu DLA suggests a decomposition of adsorbed CO on the surface (Figure S3a &S3c, Table S4).

341 In a study by Ichikawa, S. et al. (1985), they found disproportionation of CO to CO_2 and carbon

342 on a small palladium particles on a silica support.[72] They found a decrease in CO chemisorption

343 sites after seven cycles of CO adsorption and desorption at elevated temperature. At mild

344 temperature, CO disproportionation was observed by Li et al. (1991) on partially reduced cerium

345 oxide using FTIR spectroscopy.[73] Although the XPS experiment of CO exposure were made in

346 high vacuum system, the disproportionaion of CO during the CO exposure stage is not known.

347 The potential effect of this is the loss of site for CO adsorption that could lower signal (e.g. CO-

348 FTIR) for the Pd-CO isolated sites due to surface poisoning.

17

349 3.2 FTIR of Adsorbed CO

350   To probe the surface of the dilute limit alloy catalysts prepared by co-SEA, FTIR spectra were

351 obtained during and after exposure to CO. In addition, spectra of monometallic Cu and Pd catalysts

352 (at similar loadings found in the dilute limit alloy samples) were also acquired for comparison.

353 The in-situ reduction performed prior to CO-FTIR was confirmed by XPS to result in reduced

354 metal surface (see Supporting Information, Figure S2). Selected background-corrected spectra are

355 reported to show the relevant trends, with the full sets of raw spectra provided in the supporting

356 information (Figure S4). With the given dilute metal ratio, the amount of isolated atom present in

357 one cluster would depend on the total number of atoms comprising one particle. In this study, the

358 dynamic behavior of the spectra of the monometallic catalysts (Cu) was observed from the start

359 of CO introduction to the final N_2-purged spectra.

360   It should be noted that several studies have shown that exposure to CO can result in surface

361 segregation of Pd in bimetallic PdAu[71] and PdAg[74] systems. While this possibility cannot be ruled

362 out in the present case, such behavior would not alter the present conclusions, since analysis of

363 CO adsorption was performed on steady state spectra after saturation and N_2 piurging. To clearly

364 determine this would require the use of near ambient-pressure XPS and x-ray absorption

365 spectroscopy measurements that are beyond the scope of the present study.

366

367 3.2.1 Monometallic Copper and Palladium Catalysts

368   Selected spectra obtained during transient CO adsorption and N_2 purging are shown in Figure 3

369 for monometallic copper. At the onset of CO adsorption, the band intensity in the region 2070-

370 2140 cm^{-1} immediately reached its highest intensity. However, upon continuous exposure to CO,

371 the band envelope from 1900-2050 cm^{-1} gradually gains in intensity while the 2070-2140 cm^{-1}

372   region decreased slightly. This transition also produced an isosbestic point at 2070 cm^{-1}. The entire

373   band envelope became stable after 90 minutes of exposure, with two peaks becoming prominent

374   at 2050 cm^{-1} and 2010 cm^{-1} near the end. When the CO gas was purged with N$_2$, both the 2070-

375   2140 cm^{-1} region and the 2050 cm^{-1} peak intensities rapidly decreased while the 2010 cm^{-1} peak

376   increased beyond its CO-saturated intensity. Another isosbestic point at 2030 cm^{-1} was also visible

377   during this transition but not as clear due to the transient nature of the experiment. It is also evident

378   that the 1900-2000 cm^{-1} region slightly increased in intensity.

379
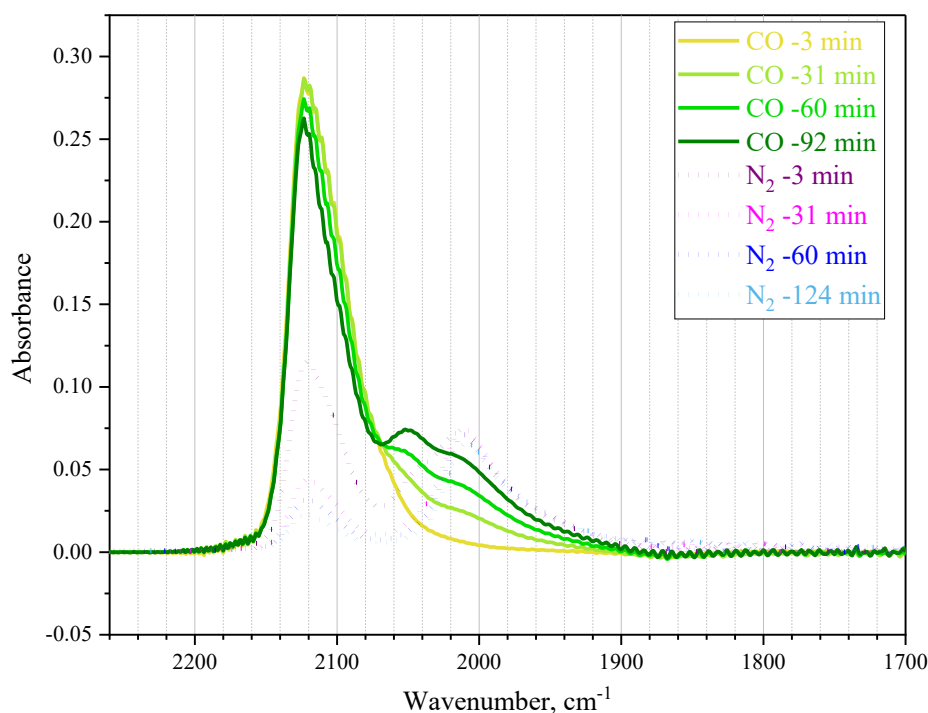
380   Figure 3. Selected blank-subtracted transient spectra for CO adsorption and desorption on monometallic

381   copper catalyst.

382   Figure 4 shows the fitted functions from the saturated CO spectra to the final N$_2$-purged spectra.

383   From these fits, the integrated areas of each associated peak were obtained and are plotted versus

384   time in Figure S5a. The total integrated intensity during the CO exposure stage shows a gradual

385   increasing trend suggesting the transformation of adsorbed species upon prolonged CO exposure.

386   Figure S5 reveals that the 2104 cm^{-1} peak decreased while the peaks in the region 1970 cm^{-1} to

387   2050 cm^{-1}increased, indicating a probable switch in adsorption mode. In the FTIR study by

388   Hadjiivanov et al. on Cu/SiO$_2$, adsorbed CO species were observed at 2045 and 2018 cm^{-1}, with

389   the former dominating at high CO partial pressure and the latter at low CO partial pressure. The

390   2045 cm^{-1} peak was assigned to the asymmetric stretch of the *gem*-dicarbonyl. The *gem*-dicarbonyl

391   species transitioned to monocarbonyl at 2018 cm^{-1}.[75] The second expected peak associated with a

392   dicarbonyl CO species was not evident in their spectra due to the possible masking of a Cu^0-CO

393   band at 2129 cm^{-1}. In the present case, during the N$_2$ purge, the 2104 and 2050 cm^{-1} peaks

394   decreased markedly, and their ratio remained approximately constant. From this trend, we assign

395   the 2104 cm^{-1} absorption as the second peak of the *gem*-dicarbonyl species together with the 2050

396   cm^{-1} peak analogous to the dicarbonyl as proposed by Hajiivanov et al. Also, the increase in

397   integrated area for the 2009 cm^{-1} peak during the nitrogen purging stage could be associated with

398   the linear CO formed from this dicarbonyl CO species. This assignment would give a ~50-55 cm^{-1}

399   difference in symmetric and asymmetric CO vibrations which is comparable to that reported in

400   the literature. [76,77]

401   The in-situ reduction at 400 °C assures that most of the copper are in a reduced state (confirmed

402   by XPS, cf. Fig S6) given that the reduction temperature peak obtained for pure copper catalyst is

403   at 210°C. Thus, the 2124 cm^{-1} peak is assigned to linear CO adsorbed on a reduced copper atom

404   (Cu^0 – CO) situated in a low index terrace site. It is noted that in the literature this peak has been

405   assigned inconsistently to Cu^0-, Cu$^{\delta+}$- or Cu$^+$-CO vibrations.[76,78–81] In the study by Scarano et al.

406   (1998) on silica-supported Cu$_2$O, they found a 2158 cm^{-1} peak at high CO coverage and proposed

407   it to be associated with the symmetric mode of a dicarbonyl species on monocoordinated Cu(I) on

408 the (111) face of Cu_2O.[76] It should be noted that the particle size under consideration is 10- to 50-

409 times higher than the particle size in this study and the adsorption temperature is at ~100 K. Thus,

410 the low intensity 2156 cm^{-1} peak can be attributed to unreduced copper in defect sites or corner

411 sites[76,82]. Bridge CO was assigned to the two low broad bands at 1981 cm^{-1} and 1879 cm^{-1},

412 specifically associated with terrace and edge/step sites of Cu, respectively. Band assignments for

413 monometallic copper are summarized in Table 2.

414
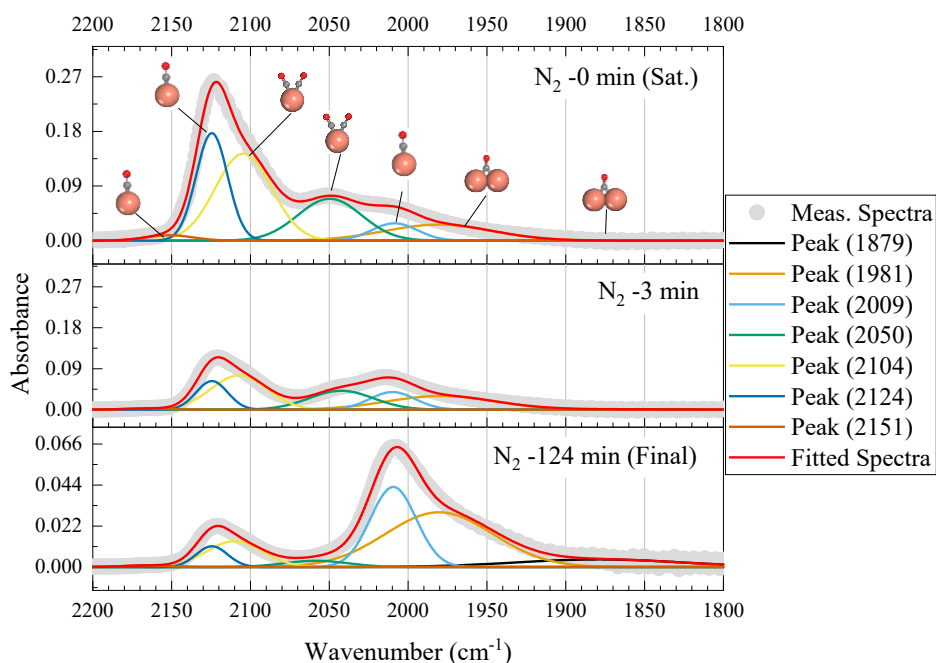
415 Figure 4. Fitted gaussian functions for 2.38%Cu/SiO$_2$ selected spectra during nitrogen purging and

416 the corresponding Cu-CO site peak assignment.

417 Table 2. FTIR CO frequency surface sites assignment for monometallic copper.

	Surface Sites								
Metal-CO Species	Corners/Defects			Terraces			Steps/Edges		
	Low Cov.	High Cov.	Ref.	Low Cov.	High Cov.	Ref.	Low Cov.	High Cov.	Ref.
Cu-(CO)$_2$							(2058, 2110)	(2050, 2104)	2045[75]

21

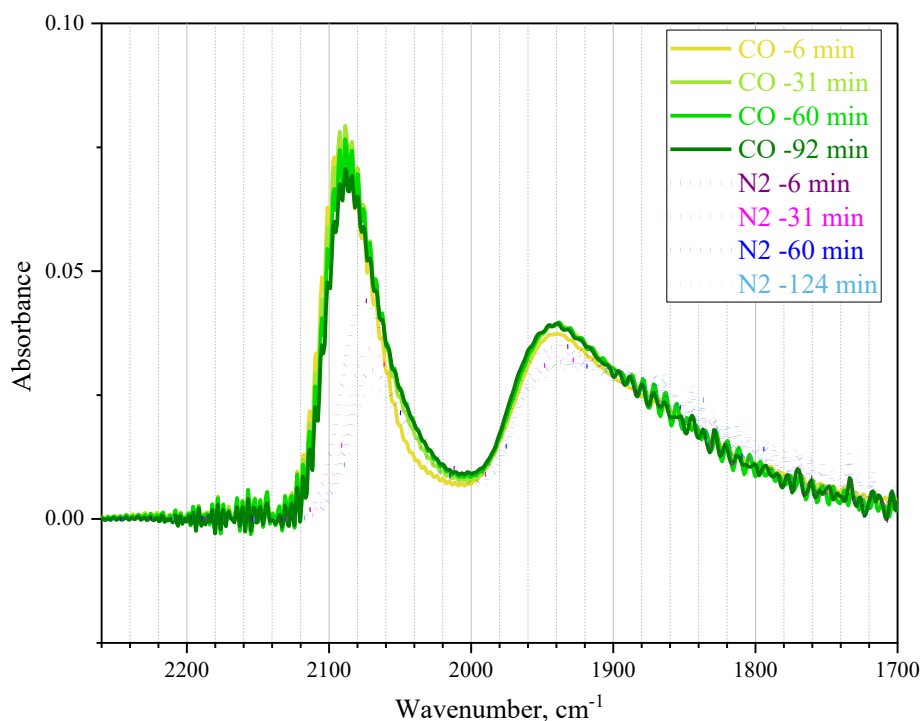| Cu-CO	2167	2151	2157[83], 2154[76]	2124	2124	2129[75]	2009	2009	2003[75]
Cu$_2$-CO				1980	1981		1879	1879	

Figure 5. Selected blank-subtracted transient spectra for CO adsorption and desorption on monometallic palladium catalyst.

Spectra obtained for 0.09%Pd/SiO$_2$ during CO adsorption and N$_2$ purging are shown in Figure 5. Band separation between linearly (>2000cm^{-1}) and non-linearly (<2000 cm^{-1}) adsorbed CO is evident and agrees well with the literature[84–87]. The asymmetric band envelope in the linear region involves a main peak at ~2086 cm^{-1} with a shoulder at ~2047 cm^{-1} becoming somewhat more prominent upon longer CO exposure. Upon purging with N$_2$, the absorbance in this region decreased but eventually reached a steady state. The non-linear region is characterized by a broad and unresolved band from 1700-2000 cm^{-1}, suggesting convolution of several peaks arising from

22

430   various highly coordinated CO surface species. At saturation, this band has a high intensity peak

431   at 1940 cm^{-1}, which decreases in intensity upon N_2 purging. Also, the shoulder at ~1800 cm^{-1} in

432   the final purge spectra shows a relatively higher intensity with respect to that obtained at saturation.

433   The fitted functions from the saturated CO spectra to the final N_2-purged spectrum for Pd/SiO$_2$

434   are shown in Figure 6. Six Gaussian functions with unique constant FWHM were obtained to fit

435   across the series of spectra. In the fitting process, two Gaussian functions were utilized for the

436   linear region and the respective wavenumbers were allowed to shift. From initial CO introduction

437   to saturation, the high wavenumber linear CO peak (2086 cm^{-1}) reached the saturation intensity

438   instantly (Figure S4b) and during the desorption stage, the peak downshifted to 2069 cm^{-1}. After

439   almost two hours of continuous nitrogen gas purging, a non-changing and lower absorption

440   intensity was obtained in both the linear and nonlinear CO vibration regions. This observation

441   suggests that palladium clusters were formed with the synthesis method at low weight loading and

442   conforms with the results obtained in previous studies.[45,88]

443   Bardshaw and Hoffman (1978) made a thorough infrared studies on CO adsorption on palladium

444   single crystal and proposed an adsorption band of a CO singleton. They associated the 2050 to

445   2120 cm^{-1} to linearly bonded CO, 1880 to 2000 to bridge CO and 1800 to 1880 to three-fold co-

446   ordination sites. Although these assignments neglected other effects such as CO coupling, the

447   same can be used to assign the adsorbed CO species for the fitted Gaussian functions (Table 3) in

448   this study. In assigning linear CO to either from a terrace or edge site, the degree of electron π-

449   backdonation to the anti-bonding orbital was used with edges sites having more available electrons

450   hence would give a relatively lower CO stretching band wavenumber.[89]

451

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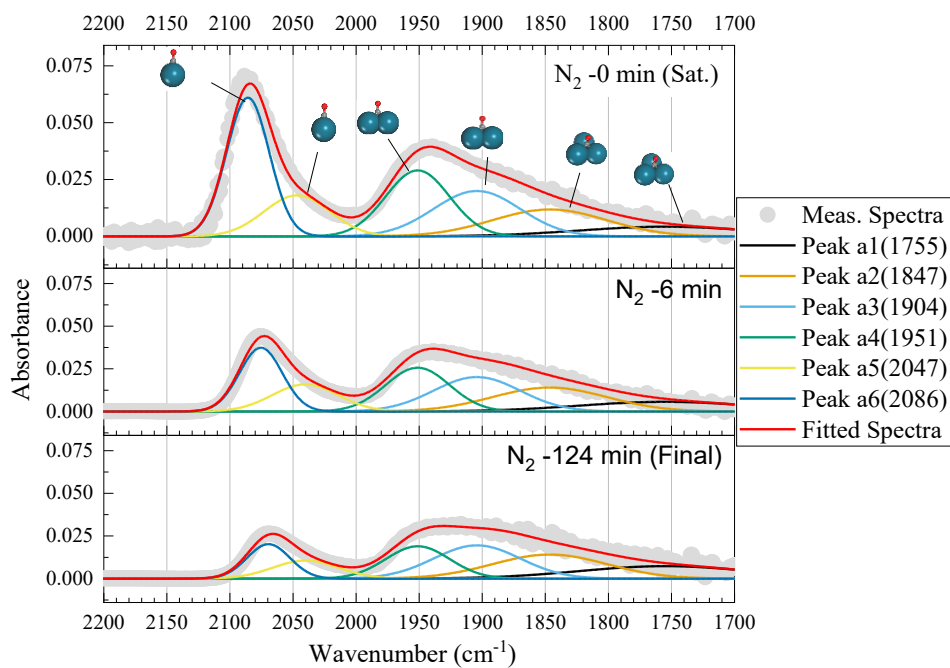
Figure 6. Fitted Gaussian functions for 0.09%Pd/SiO$_2$ spectra taken after 124 minutes of nitrogen

purging with the corresponding Pd-CO site peak assignment.

Table 3. FTIR CO frequency and surface site assignments for monometallic palladium.

Metal-CO Species	Surface Sites					
	Terraces			Steps/Edges		
	Low Cov.	High Cov.	Ref.	Low Cov.	High Cov.	Ref.
Pd-CO	2069	2086	2095[88], 2092[90], 2077[91], 2073[92]	2038	2047	2050[91], 2049[13], 2033[9]
Pd$_2$-CO	1949	1951	1949[90], 1951[92], 1937[9], 1956[88]	1904	1904	1914[92]

Pd$_3$-CO	1847	1847	1836[90], 1840[91], 1830[88]
	1755	1755	

456

3.2.2 Pd-Cu Dilute Limit Alloy

457

458 Transient spectra for the 1/19 (Figure S6a), 1/30 (Figure S6b) and 1/74 (Figure S6c) dilute limit

459 alloy catalysts were acquired during CO adsorption and N$_2$ purging with the selected blank-

460 subtracted spectra shown in Figure 7a-c. Overall, the progression of the spectra is similar to that

461 obtained for monometallic copper. Two prominent features at ~2050 and ~2010 cm^{-1} are observed,

462 together with a gradual decrease in ~2100 cm^{-1} intensity as CO reaches saturation. There is a rapid

463 decrease in intensity in the ~2100 cm^{-1} region and 2050 cm^{-1} region together with an increase in

464 2010 cm^{-1} region during the N$_2$ purging step. Overall, the results suggest that the nanoparticle

465 surfaces are largely made up of Cu sites, as expected for a dilute limit alloy. In this category of

466 isolated atom on a ultrasmall cluster of copper particle, the expected number of surface isolated

467 atoms for a diluted bimetallic would be small. Unlike the single atom alloy produced by the

468 galvanic displacement approach, the abundant base metal starts at a particle size with a significant

469 number of total atoms. The relatively large nanoparticles with several isolated atoms on the surface

470 may segrate once exposed to CO as presented by van Spronsen, M. A., et al. (2019) on a

471 Pd/Ag(111) system.[74] CO induced segration of surface atoms are also observed in moderate metal

472 ratio alloyed bimetallic catalysts and alloyed core-shell structure. [70,71] This state of segration is

473 believed to be not occurring in the current dilute limit alloy Pd-Cu catalysts prepared by co-SEA

474 with small size abundant metal.

475 The clearest comparison between monometallic Cu and the dilute limit alloys can be made by

476 considering the most strongly adsorbed species on each sample. Figure 8 shows the normalized (to

477   the ~2010 feature) final N_2-purged raw spectra for each DLA sample and for monometallic copper.

478   Once again, the overall similarity in the absorbance shape of these spectra is noted. However, two

479   regions (2000-2100 cm^{-1}, 1830-1940 cm^{-1}, highlighted in red) clearly show an added CO

480   absorbance that is absent from monometallic copper. These contributions to the spectra can be

481   attributed to the presence of Pd as either DLA sites or monometallic Pd.

482   Figure 9a shows the fitted spectra for monometallic 0.09 wt% Pd/SiO_2 (top) and 2.38 wt%

483   Cu/SiO_2 (bottom), and the most dilute Pd-Cu catalyst (middle). Deconvolution of DLA Pd-Cu

484   catalyst made use of the fitted Gaussian function parameters obtained from monometallic Cu

485   (green peaks). During fitting, the FWHM and peak positions of the Cu peaks were kept constant

486   while some peak ratios (minimal) and the overall intensity height were allowed to change. As can

487   be seen, there are no obvious features in the DLA that can be associated with monometallic Pd

488   (blue peaks). In order to fit the remaining spectral features, three new Gaussian peaks were

489   required. As seen more clearly in Figure 9b, a pair of linear CO peaks at 2046 and 2021 cm^{-1} were

490   found, with a single lower-frequency feature at 1908 cm^{-1} in the bridging region. Table S5 gives a

491   summary of Gaussian function parameters obtained during the fitting. It should be noted that these

492   three new peaks are present to varying extents in the other dilute limit alloy samples as shown in

493   Figure S7. In a study using Pd-Cu SAA nanoparticles with a particle size range from 3 to 7 nm, an

494   isolated peak at 2033 cm^{-1} associated with the linear CO on isolated Pd atom on copper was

495   observed. [9] In this study by Shan et al. (2020), the monometallic copper has no adsorption on the

496   2000 cm^{-1} region as compared with the current study. One possible explanation for this is the

497   significant difference in particle size. The SEA-derived catalysts explored here exhibit a Cu

498   particle size less than 1.5 nm, which high proportion of edge/step sites. In contrast, the catalysts

499   in Shan et al. (2020) ranged from 3-7 nm, which comprises mainly terrace sites. The electron

500   density on the copper surface is also dependent on the particle size as well as there possibly being

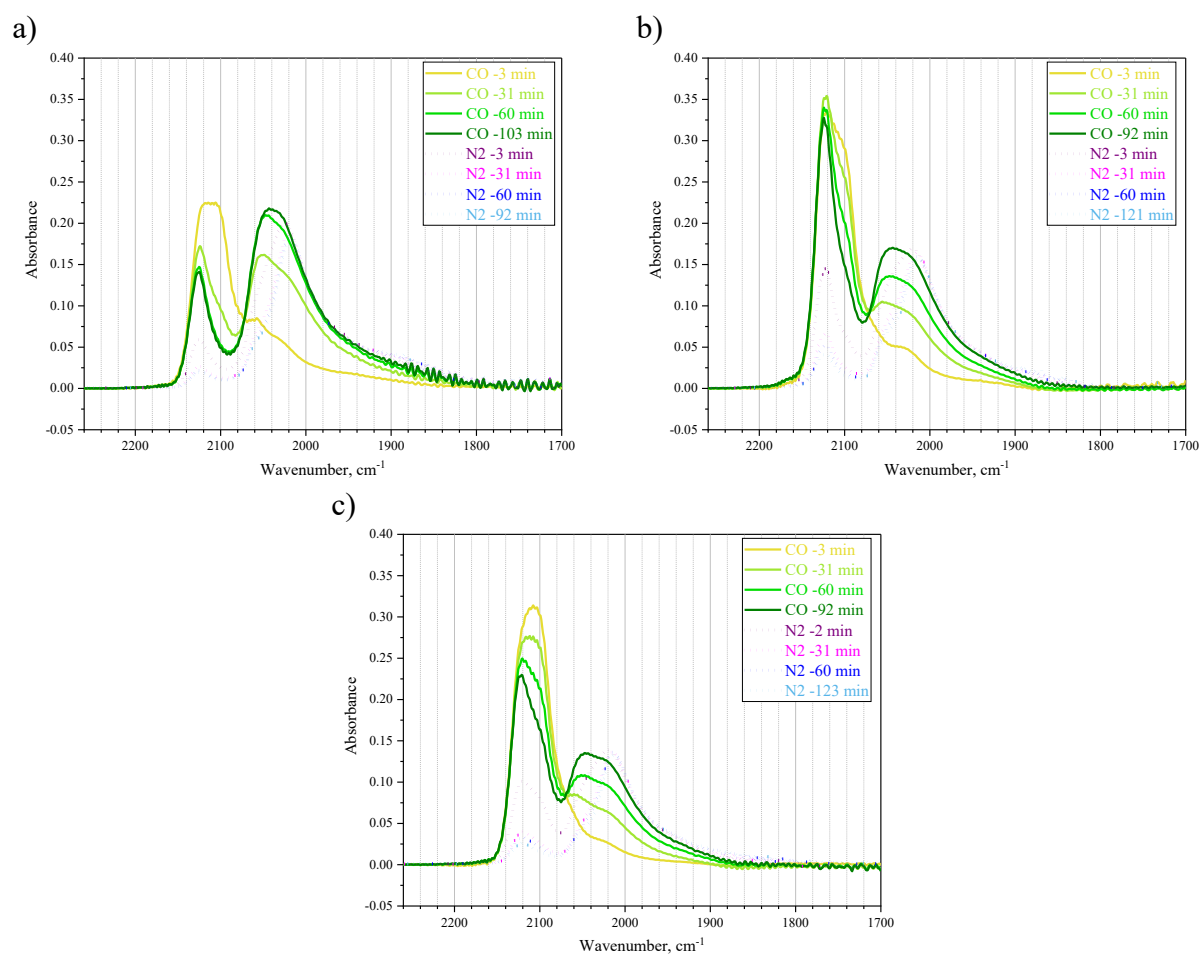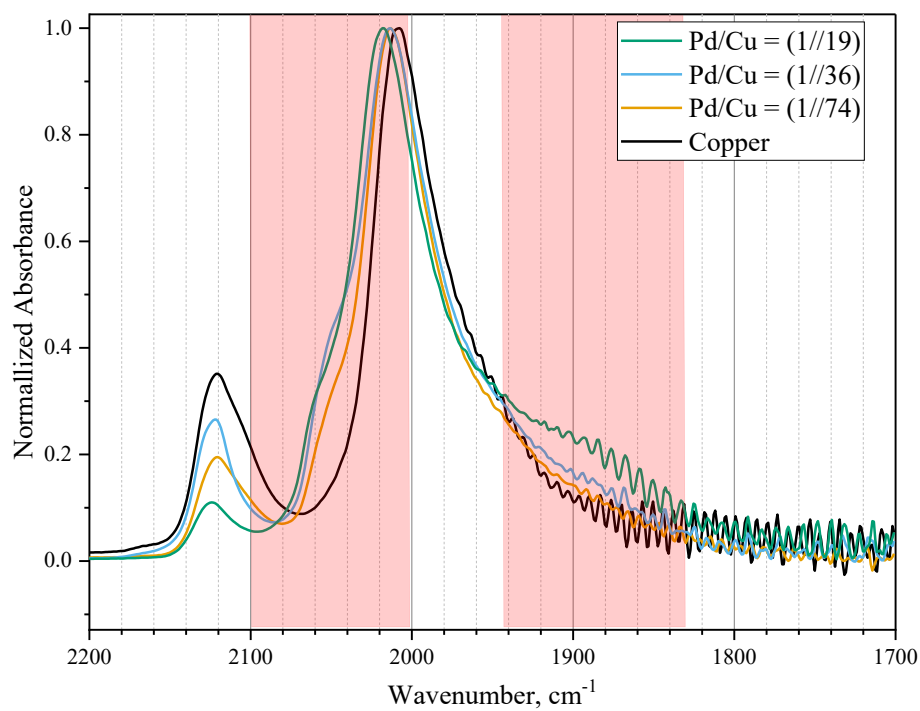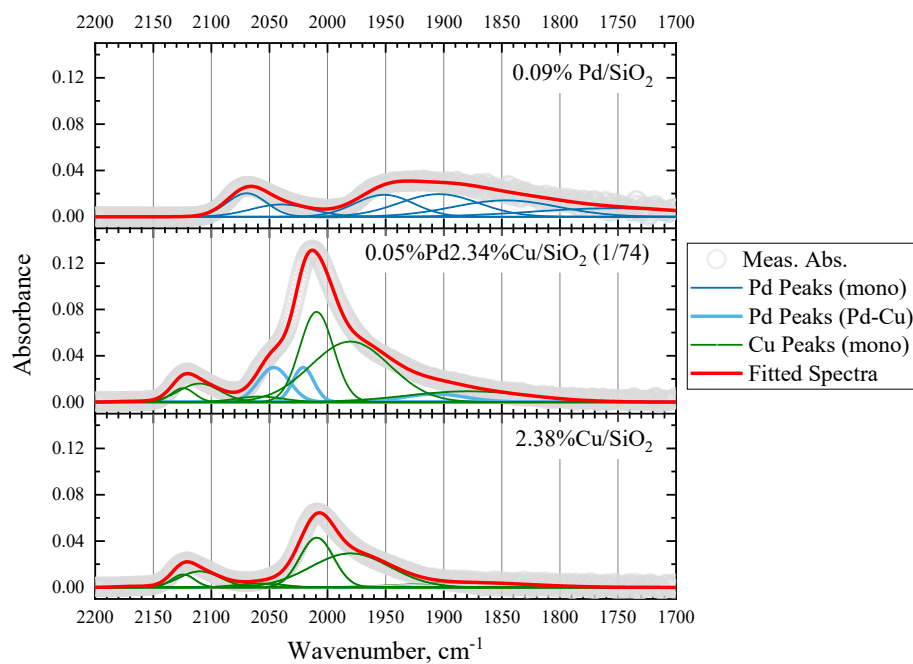501   a stronger metal-support interaction.

502

Figure 7. Selected blank-subtracted spectra for (a) 1/19, (b) 1/36 and (c) 1/74 Pd dilution on Cu.
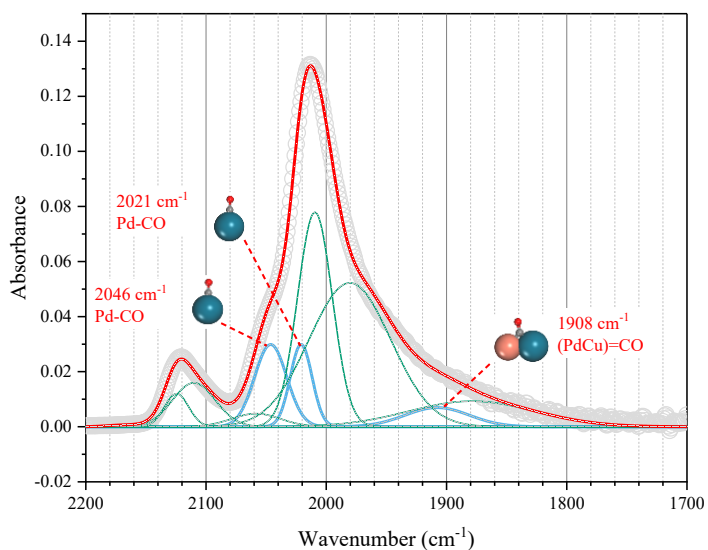
504

505

Figure 8. Normalized CO-FTIR spectra taken after 124 minutes of nitrogen purging for monometallic copper and dilute limit alloy series Pd-Cu.

508   (a)

509

510   (b)

511

512   Figure 9. (a) Fitted spectra (final N_2 purged) for monometallic Pd and Cu, and bimetallic Pd-Cu

513   system and (b) three additional peak assignment for isolated palladium atom sites on Pd-Cu dilute

514   limit alloy.

515

516   3.3 DFT Calculation Results

517   DFT simulations were employed to assist in the band assignments of the three new peaks

518   observed in the DLA samples as well as for the monometallic palladium and copper catalysts. Low

519   coverage calculations were performed for four surface models (111, 100, 211 and 110) of

520   monometallic palladium and copper surfaces and single atom alloy surfaces of bimetallic

521   palladium-copper systems. Figure 10 and S4 illustrate the surface models and CO adsorption sites.

522   Table 4 shows the calculated wavenumber for adsorbed CO species on the various sites for the

523   monometallic copper and palladium systems with the corresponding observed experimental CO-

524   FTIR peaks. In all configurations, linearly adsorbed CO shows a wavenumber greater than 2000

525   cm^{-1}. The calculated wavenumbers for linearly adsorbed CO does not go beyond 2102 cm^{-1} (which

526   we compute for a Cu adatom on Cu(111)). This conforms with the experimental results for

527   monometallic palladium. For copper, the measured peak at 2124 cm^{-1} was assigned to a CO

528   adsorbed linearly atop a Cu-adatom on the Cu(111) surface facet (Table 4). The DFT calculated

529   wavenumber (2102 cm^{-1}) close to this value suggests a linear CO adsorbed on an undercoordinated

530   copper adatom situated over a low index terrace site (although all Cu-adatom models display

531   somewhat similar frequencies – Table 4).

532   DFT calculations also confirmed the presence of gem-carbonyl species where a $Cu(CO)_2$

533   complex was predicted to form over Cu(111) surface facets. The *gem*-dicarbonyl was oriented

534   such that the (C-Cu-C) angle is 118.83° (Figure S8a & S8b). For the asymmetric stretching, we

535   computed a frequency of 2064 cm^{-1} versus 2095 cm^{-1} for the symmetric stretching (Table 4)[93]. The

536   difference of the calculated dicarbonyl wavenumbers (31cm^{-1}) is somewhat lower than that of the

537   experimental observation ($52 cm^{-1}$); however, it is difficult to determine if this difference originates

538   from computational uncertainties or the presence of another Cu site not investigated in this study.

539   For terrace surface facets, the calculated wavenumber for linear CO adsorbed on Cu(100) is

540   greater than that of Cu (111) site, which implies a lesser degree of π-electron back donation on

541   Cu(100). For copper, the degree of π backdonation is in direct correlation with the coordination

542   number of the adsorption site with the fcc(111) surface atoms having CN=9 as compared with

543   fcc(100) with CN=8. However, for monometallic palladium, the degree of π backdonation is lesser

544   with increasing coordination number. This suggests that sharing of available electrons is

545   dominating in palladium surface atoms. The higher coordination number of Pd (111) terrace sites

546   as compared with Pd (100) shows a higher wavenumber, which implies less available electrons for

547   π backdonation. With respect to the number of metal atoms coordinated with the adsorbed CO, the

548   calculated frequency obeys the trend as atop > bridge > 3-fold-hollow > 4-fold-hollow.

549   Table 5 shows the calculated frequencies for CO adsorbed on isolated palladium on four types

550   of copper surface facets with an embedded and ad-atom configuration (Figure 10 & 12). From the

551   experimental CO vibration results for the isolated Pd, the association with the calculated

552   wavenumbers was made accordingly. For a linearly adsorbed CO on an isolated palladium atom,

553   a wavenumber of $2052 \ cm^{-1}$ was calculated for atop CO on Pd embedded on a Cu(100) facet

554   (Figure 10e). Within the range of computation certainty, vibration frequencies for CO adsorbed on

555   Pd adatoms on Cu(100) ($2052 \ cm^{-1}$), Cu(111) ($2058 \ cm^{-1}$) and Cu(211) ($2048 \ cm^{-1}$) were also

556   calculated (Table 5). A significantly lower wavenumber at $2026 \ cm^{-1}$ was obtained for Pd single-

557   atoms embedded in a Cu(111) facet (Figure 10c). The trend in CO frequency can be explained by

558   the Dewar–Chatt–Duncanson (DCD) model.[94] With copper acting as a ligand that donates

559   electrons to the isolated palladium atom, as the coordination number of Pd with Cu increases, the

560   number of electrons available for backdonation also increases giving a much lower CO

561   wavenumber. DFT calculated a 1911 cm^{-1} frequency on CO forming a bridge between Pd and Cu

562   on a Pd embedded on a Cu(211) terrace site. This frequency is within the region of the additional

563   bridge peak that was required to fit the spectra for the Pd-Cu DLA (1908 cm^{-1}). As the particle

564   gets smaller, the percentage of undercoordinated atoms such as, e.g. edges and steps, increases

565   relative to terrace surface atoms. The Benfield (1992) model on cuboctahedral cluster particles

566   suggests that as the particle size falls below 2 nm the fraction of edge atoms can go beyond 50%.[37]

567   Given this surface atom distribution, there is a high probability of this isolated Pd atom situated

568   on step-edge sites of the abundant metal. For clusters, geometrical factors could change, such as a

569   shorter span of the terrace facet making no significant difference in (111) and (100) facets CO

570   adsorption.

571

572 Table 4. DFT calculation on CO vibration frequency for the monometallic copper and palladium

573 systems. If the adsorption site is not located at a potential energy minimum, it is considered

574 unstable and hence no frequency was calculated. Figure S8 & S9 lists all the adsorption sites for

575 the monometallic systems.

| System	Adsorption Site	C-O stretching frequency (cm^{-1})			
		Copper		Palladium	
		Calculated	Observed (range)	Calculated	Observed (range)
(100)	atop (a)	2052	2050 to 2058	2057	2038 to 2047
	bridge (b)	1913		1890	
	4-fold-hollow (4f)	1727		1756	1755
(110)	atop (a)	2061	2050 to 2058	2050	2038 to 2047
	terrace bridge (tb)	1931		1822	
	step bridge (sb)	1928		1881	
(111)	atop (a)	2017	2009	2072	2069 to 2086
	Bridge (b)	1853	1879	1882	
	fcc, hcp	1855	1879	1801	
(211)	atop step (ts)	2064	2050 to 2058	2054	2038 to 2047
	atop terrace (tt)	2038		unstable	
	step bridge (eb)	unstable		1893	1904
	step-terrace bridge (sb1)	unstable		unstable	
	step-terrace bridge (sb2)	1915		1873	
	3-fold-hollow (h)	1828		1725	1755
	3-fold hollow (f1)	1802		1794	1755
Cu-adatom /Cu(111)	$Cu(CO)_1$	2102	2124	NA	
	$Cu(CO)_2$	2064,2095	(2050 to 2058, 2104 to 2110)	NA	
Cu-adatom /Cu(211)	$Cu(CO)_1$	2097		NA	
	$Cu(CO)_2$	2037,2074		NA	
Cu-adatom /Cu(110)	$Cu(CO)_1$	2084		NA	
Cu-adatom /Cu(100)	$Cu(CO)_1$	2084		NA	
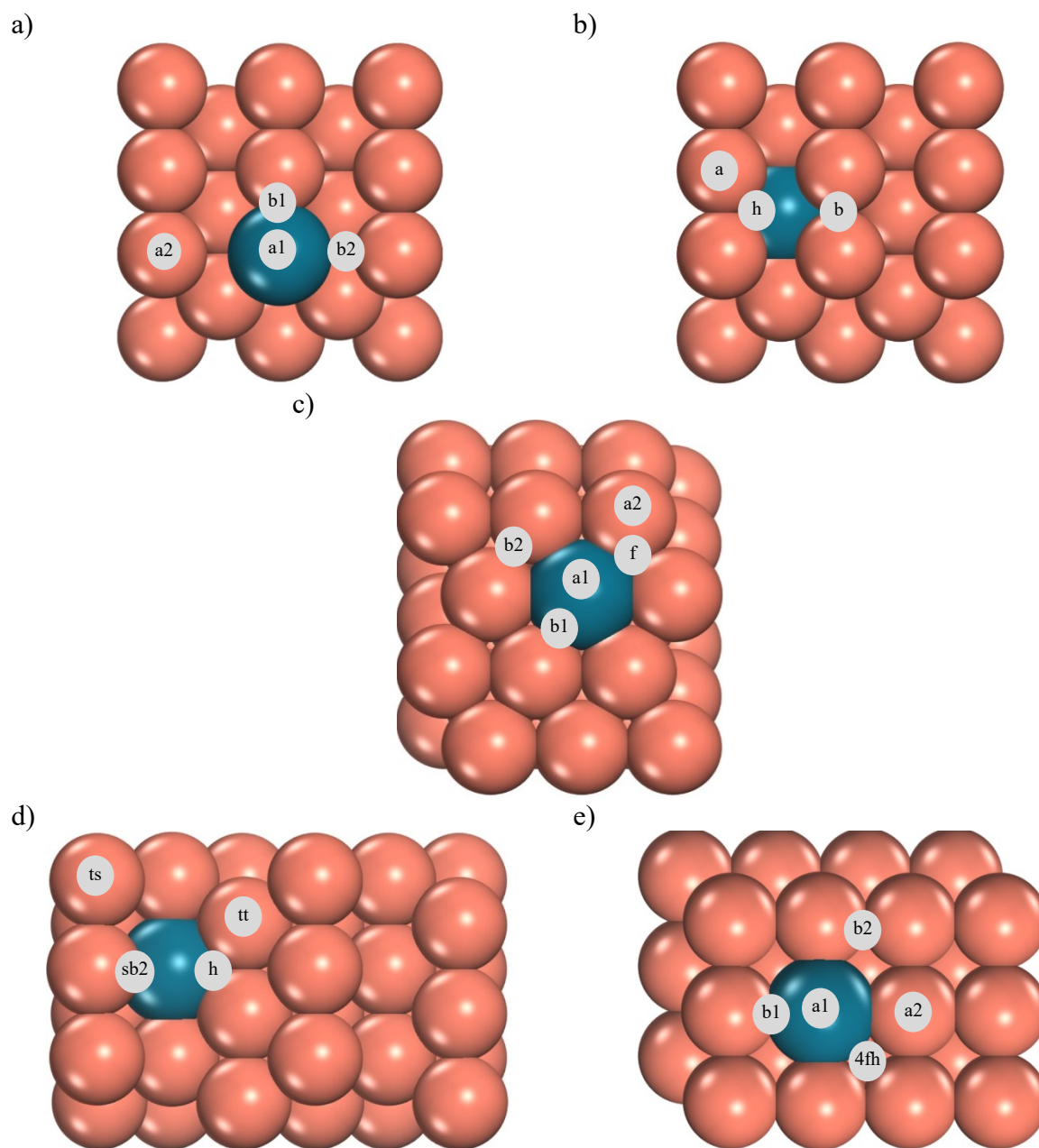
Figure 10. Stable adsorption sites for Pd-Cu bimetallic catalyst. a) Cu(110) facet alloyed with a Pd atom replacing a step Cu atom, b) Cu(110) facet alloyed with a Pd atom replacing a terrace Cu atom, c) Cu(111) facet alloyed with a Pd atom (Pd/Cu111), d) Cu(211) facet alloyed with a Pd atom (Pd/Cu211) where the Pd atom has a coordination number (CN) of 10, and e) Cu(100) facet alloyed with a Pd atom (Pd/Cu100).

582 Table 5. DFT calculation on CO vibration frequency for the dilute limit alloy bimetallic palladium-

583 copper system.

| System | Adsorption Site | C-O stretching frequency (cm^{-1}) | |
		DFT Calculated	Observed (Pd)
Pd/Cu(100)	atop-Pd (a1)	2052	2046
	atop-Cu (a2)	1928	
	Pd-Cu bridge (b1)	1930	
	Cu-Cu bridge (b2)	1914	
	hollow (4-fold)	1927	
Pd/Cu(111)	atop-Pd (a1)	2026	2021
	atop-Cu (a2)	2057	
	Pd-Cu bridge (b1)	1927	
	Cu-Cu bridge (b2)	1865	
	2Cu-1Pd-FCC (F)	1873	
Pd/Cu(211)	3-fold-hollow (h)	1852	
	Pd-Cu step-terrace bridge (sb2)	1911	1908
	atop-Cu step (ts)	2066	
	Atop-Cu terrace (tt)	2044	
Pd/Cu(110) (Pd in step)	atop-Pd (a1)	2046	
	atop-Cu (a2)	2061	
	2-step bridge Pd-Cu (b2)	1882	
	1-step bridge Pd-Cu (b1)	1934	
Pd/Cu(110) (Pd in terrace)	atop-Cu (a)	2063	
	bridge-Cu-Cu (b)	1932	
	3-fold-hollow (h)	1822	
Pd-adatom/Cu(100)	atop-Pd	2052	2046
Pd-adatom/Cu(111)	atop-Pd	2059	2046
Pd-adatom/Cu(211)	atop-Pd	2048	2046

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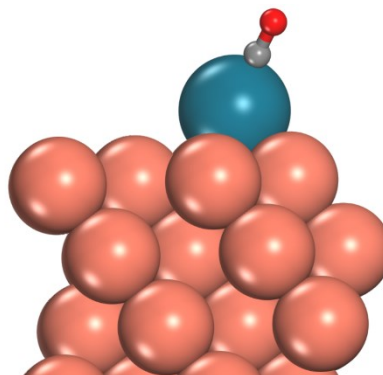
586

a)
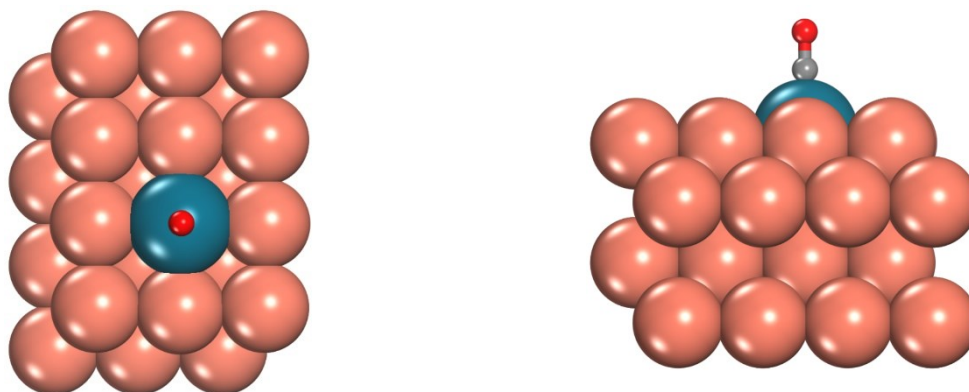
b)
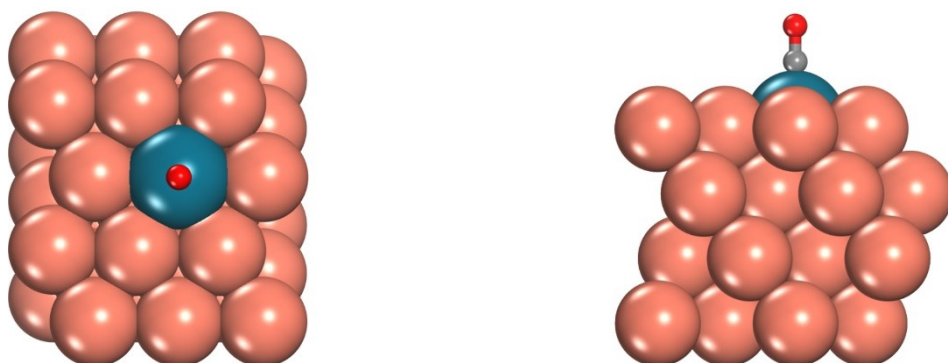
c)
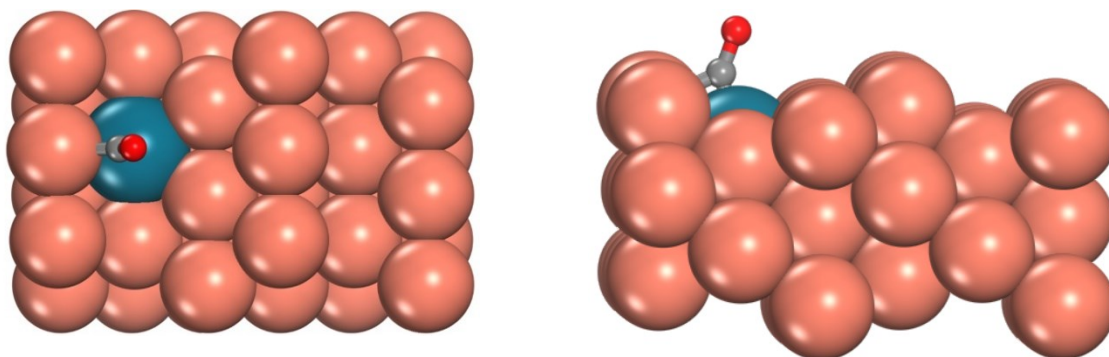
587   Figure 11. Top and side views of surface models corresponding to experimentally observed CO

588   vibration; (a) 2052 cm^{-1} atop CO on Pd/Cu(100), (b) 2026 cm^{-1} atop CO on Pd/Cu(111), and (c)
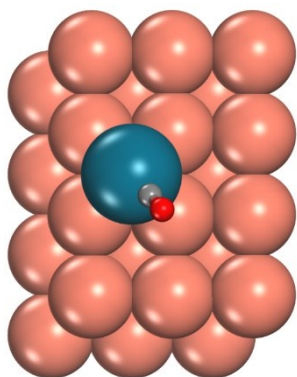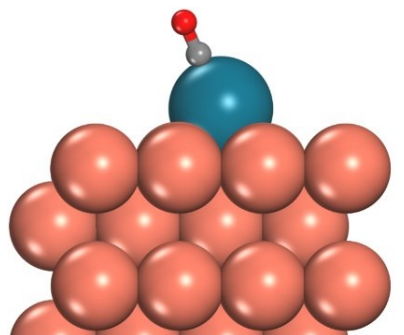
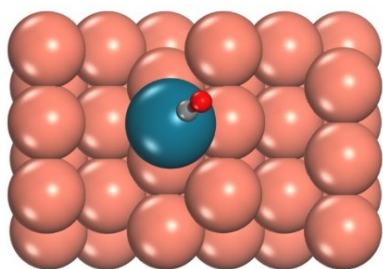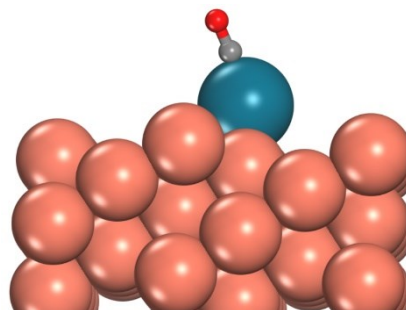589   1911 cm^{-1} bridge CO.

a)

b)

c)

d)

e)

f)
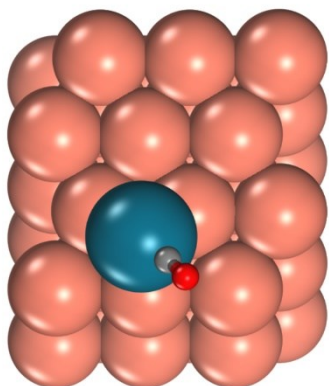
591   Figure 12.Top and side views of surface models of Pd-adatom with CO vibrations;(a,b) 2059 cm⁻

592   ^{1} on Cu(111),(c,d) 2052 cm^{-1} on Cu(100), and (e,f) 2048 cm^{-1} on Cu(211) .

593

594

595

4. Conclusions

A facile synthesis method to prepare DLAs of Pd in Cu supported on SiO_2 has been demonstrated. Simultaneous co-SEA combined with controlled removal of ligands and reduction of the metal resulted in the formation of small metallic clusters on the support surface. These reduced particles presented isolated Pd sites on their surface, as confirmed by in-situ FTIR spectroscopy measurements of adsorbed CO and complementary DFT computations. In addition, at the low dilution levels examined, no evidence could be found for formation of monometallic clusters or particles of Pd.

The present results suggest that co-SEA can provide a more general and scalable approach to producing highly dispersed DLA catalysts as compared with other more exotic approaches. While only Pd and Cu are considered here, SEA is applicable to the entire range of transition metals relevant to catalysis. In the present case, Pd was found to situate itself on the Cu surface, or at least be drawn out to the surface during CO adsorption. However, depending on the metals in question, a dilute species might be submerged within the other metal. Ongoing studies in our laboratory are exploring these Pd-Cu and other co-SEA-derived DLA catalysts for a variety of catalytic applications, including selective hydrogenation reactions.

5. Supporting Information

Supporting raw CO-FTIR spectra, STEM images, fitted Gaussian functions parameters, additional DFT models and XPS spectra are available in the supplementary material.

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