

Stress concentration of a micro-void embedded in a bi-layered material considering the boundary effects

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ABSTRACT

This paper extends the Eshelby’s problem of one inhomogeneity embedded in a homogeneous infinite domain to a bi-material infinite domain. The equivalent inclusion method (EIM) is used to simulate the inhomogeneity by an inclusion with a polynomial eigenstrain. The fundamental solution of a point force in a bi-material is used to formulate the domain integral over the inclusion. For a finite bi-material domain, the boundary integral equation (BIE) takes into account the boundary responses by a single domain instead of utilizing the conventional multi-region BIE scheme. The EIM can similarly be used, and the elastic field can be obtained with tailorable accuracy based on the order of the polynomial eigenstrain. The algorithm is particularly suitable to simulate a defect in thin film/substrate systems or other similar bi-layered materials. Particularly, the stress concentration of a microvoid embedded in a bi-layered solar panel is investigated. The size and location of the void referred to the interface exhibits considerable effects on the stress concentration factor. Numerical case studies demonstrate the effectiveness and accuracy of the algorithm, and parametric studies show the boundary effects on the stress concentration of a microvoid in a finite bi-material under a uniform far field strain.

25 **INTRODUCTION**

26 The multi-layered systems have been widely utilized in versatile engineering and construc-
27 tion aspects, such as asphalt pavements (Yin and Prieto-Muñoz 2013), thin film surface coatings
28 with protective and functional purposes (Abu-Thabit 2020; Ruys and Sutton 2021), and composite
29 laminates (Anbusagar et al. 2015; Rana and Fangueiro 2016). However, the defects during manu-
30 facturing process, such as air voids, may significantly jeopardize the reliability and lifetime of the
31 overall bi-material system. Sengab and Talreja (Sengab and Talreja 2016) summarized two main
32 sources of those defects, (i) impurities and air evaporating during curing process; (ii) entrapment
33 of air during manufacturing process. For example, in solar panel manufacturing (Yin et al. 2022b),
34 any voids in a solar panel may disturb the light transmission, form hot spots under strong sunlight,
35 and cause microcracks and failure of the solar panel. Even for a homogeneous encapsulate layer,
36 the defects can cause stress concentrations leading to cracks and failure. Especially, when defects
37 are close to the interface S , the discontinuity of stress across the interface augments the stress
38 concentration and singularity effects. Therefore, high fidelity stress analysis may provide more
39 insights to understand this phenomena.

40 To investigate the stress transfer between layered materials, Stoney (Stoney 1909) proposed an
41 approach with plate system assumptions, such as thick substrate to ignore bending stiffness of thin
42 film, equal twist curvatures and spatially constant surviving stress (Ngo et al. 2007). Because the
43 strong assumptions violate practical applications, several subsequent extensions have been proposed
44 to relax them (Wikström et al. 1999; Park and Suresh 2000). However, the Stoney theory ignores
45 the shear stress transfer, modified theories (Haftbaradaran et al. 2012; Zhang et al. 2021) have been
46 proposed to consider the interfacial sliding effects. Since the above previous works assume two
47 dissimilar homogeneous material phases, therefore those models cannot provide accurate analysis
48 for bi-material system with defects. Regarding influence brought by micro defects, Katnam et al.
49 (Katnam et al. 2011) investigates the formulation of air voids with two adhesive mixing techniques
50 and used X-ray to detect and evaluate porosity; Omairey et al. (Omairey et al. 2021) summarized
several failure modes of adhesive joints of composites, where adhesive defects and substrate defects

52 can cause high stress concentration leading to failures; Mishnaevsky ([Mishnaevsky 2022](#)) reported
53 that even the defects in adhesive of the wind turbine blade may not dramatically disturb overall
54 stress field, the local high stress concentration will lead to crack initiation.

55 To understand the effects of defects, the equivalent inclusion method (EIM) ([Eshelby 1957](#);
56 [Eshelby 1959](#)) was proposed to replace the defects with same matrix material along with inelastic
57 strain, eigenstrain, to be determined by equivalent stress equations. With solved eigenstrain, the
58 elastic fields can be acquired through superposition of initial fields and disturbance of eigenstrain,
59 which is domain integral of fundamental solution over the inhomogeneity. Thanks to the versatility
60 of fundamental solutions, EIM has been widely extended to other problems, such as heat conduc-
61 tion ([Hatta and Taya 1986](#); [Wu et al. 2021](#)), dynamic elasticity ([Song and Yin 2018](#)), etc. The
62 Eshelby's solution of one inhomogeneity over the infinite matrix ignores the interactions among
63 inhomogeneities themselves and the boundary ([Liu and Yin 2014](#); [Wu and Yin 2021](#)). Based
64 on EIM, pioneers developed micromechanical models, such as the dilute, Mori-Tanaka([Mori and](#)
65 [Tanaka 1973](#); [Kanit et al. 2003](#); [Yin and Zhao 2016](#)) and self-consistent models ([Hershey 1954](#);
66 [Kroner 1958](#)), which provides effective mechanical properties of composites and bridge the mi-
67 crostructure and macroscopic behaviors. Other contributions on the homogenization schemes from
68 linear elasticity to nonlinear rate-dependent problems can be found in ([Zaoui 2002](#)).

69 In the literature, several previous works explore the stress intensity factors (SIFs) caused by
70 cracks and interfacial defects. Rather than using the conventional FEM, Treifi and Oyadiji ([Treifi and](#)
71 [Oyadiji 2013](#)) proposed a fractal-like FEM to investigate SIFs of notch bodies with displacement
72 interpolation functions. Bouhala et al. ([Bouhala et al. 2013](#)) developed crack-tip enrichment
73 functions with extended FEM (XFEM) to study SIFs for cracks terminating at interface of bi-
74 material; Pathak et al. ([Pathak et al. 2011](#)) combined element free Galerkin method and XFEM
75 on crack interaction problems. Kaddouri et al. ([Kaddouri et al. 2006](#)) studied a practical case with
76 couple metal-ceramic on factors associated with perpendicular cracks to bi-material interface, **such**
77 **as** distance of crack-tip to the interface.

78 As for **boundary element method (BEM)** with Kelvin's solution, the multi-region scheme is

commonly used that interface of inhomogeneity and interface of bi-material require surface mesh and continuity equations are built to formulate the boundary value problem (BVP) (Beer et al. 2008; Liu et al. 2011). Fortunately, the continuity equations on interfaces of bi-material or multi-layered material can be mathematically considered with the fundamental solutions. Walpole (Walpole 1996) derived fundamental solutions to two-jointed dissimilar isotropic half-spaces through method of images; Yue (Yue 1995) proposed Yue's treatment, which is a generalized Kelvin's solution to multi-layered material; and other contributions in the literature can be found in review (Liu et al. 2011). Xiao et al. (Xiao et al. 2019) applied Yue's treatment (Yue 2015) in BEM to investigate semi-infinite transversely isotropic domain, especially for simulation of rocks. For a bi-material system, Yue's treatment can be reduced to explicit formulae, and Wu et al. (Wu et al. 2022) completed Walpole's solution and applied it for analysis of bi-material system. The above works, though, save efforts in discretizing bi-material interface, when the number of inhomogeneity increases, dimension scale varies or close to boundary, the multi-region scheme on inhomogeneity requires both considerable computational resources and preparation process.

In our recent work (Yin et al. 2022a), the algorithm of inclusion-based boundary element method (iBEM) is designed to handle drastically increase of DOFs in simulation of composites. Using technique of fundamental solution, the material mismatch between inhomogeneity and matrix can be simulated with eigenstrain field without mesh of subdomains. Compared with Eshelby's uniform eigenstrain assumption in (Eshelby 1957), eigenstrain is presented by Taylor series expansion at the centroid of inhomogeneity (Mura 1987). The algorithm iBEM combines the BEM and EIM, where the boundary effects and interactions between inhomogeneities are considered in BEM global matrix and equivalent stress equations, respectively. Solving the system of linear equations, the boundary responses and eigenstrain field can be obtained. The advantages of iBEM are: (i) for each inhomogeneity, the number of DOFs is fixed, 6, 24, 60 for uniform, linear and quadratic order, respectively; (ii) avoid any subdomain mesh, including bi-material interface and inhomogeneity and potential numerical errors brought by them; (iii) the merit of BEM and fundamental solution is retained that internal fields are expressed in boundary and domain integrals.

This paper aims to perform elastic analysis of an inhomogeneity embedded a bi-material system through the single-domain iBEM implemented with bi-material fundamental solution. In the following, the problem of a bi-material system with an inhomogeneity is firstly proposed, and then the fundamental solutions of bi-material, domain integrals over spherical inhomogeneity and global matrix of iBEM are introduced. Subsequently, the aforementioned iBEM is verified with FEM for a benchmark comparison. Applying the solution to a solar module containing a glass layer over a concrete panel, when a microvoid is embedded in the substrate, the SIFs are investigated with various distance to the bi-material interface. Finally, some conclusive remarks are discussed.

PROBLEM STATEMENT

Consider a domain \mathcal{D} embedded with one subdomain Ω_1 is composed of two dissimilar isotropic domain, where the upper phase \mathcal{D}^+ and the lower phase \mathcal{D}^- generally exhibits different material properties C' and C'' , respectively. For instance, the stiffness tensor of \mathcal{D}^+ is $C'_{ijmn} = \lambda' \delta_{ij} \delta_{mn} + \mu' (\delta_{im} \delta_{jn} + \delta_{in} \delta_{jm})$, where λ' and μ' are lame constants of \mathcal{D}^+ . The dimensions of the bi-material system are defined in Fig.1 that, (i) T_1 and T_2 are thickness of \mathcal{D}^+ and \mathcal{D}^- , respectively; (ii) l and b are length and width. Shown in Fig. 1, \mathcal{D} is subjected to prescribed boundary conditions, where t_i and u_i represents surface traction and displacement, respectively. Without the loss of any generality, the bi-material interface S is chosen as parallel to plane $x_1 - x_2$ at $x_3 = 0$. In the following, two assumptions are made: (i) the embedded subdomain is filled with an isotropic material and its stiffness tensor C^I_{ijmn} ; (ii) the bi-material and subdomain interfaces are perfect without any debonding behavior, which satisfy the continuity equations on both displacement and normal traction shown in Eq. (1).

$$u_i(\mathbf{x}^+) = u_i(\mathbf{x}^-), \quad \sigma_{ij}(\mathbf{x}^+) \mathbf{n}_j(\mathbf{x}^+) = \sigma_{ij}(\mathbf{x}^-) \mathbf{n}_j(\mathbf{x}^-) \quad (1)$$

where "+" and "-" represents the inward and outward side of the bi-material interface S or subdomain interface, respectively; \mathbf{n} is the unit surface normal vector. Subsequently, the BVP can be formulated and it can be solved through the conventional multi-region scheme with interface mesh, which

131 commonly demands high computational costs due to the singularity and discontinuity along the
 132 interface. This paper proposes an alternative method in which the fundamental solution for a bi-
 133 material infinite domain is directly applied to Eshelby's equivalent inclusion method and boundary
 134 integral method.

135 FORMULATION

136 Fundamental Solutions

137 Considering a two-jointed dissimilar half-spaces, the displacements at field point \mathbf{x} can be
 138 expressed through the superposition of Kelvin's solution and image terms (Walpole 1996). The
 139 Green's function define the displacement response of any field point \mathbf{x} caused by unit excitation at
 140 source point \mathbf{x}' . Given a unit concentrated force $f_j(\mathbf{x}') = n_j \delta(\mathbf{x}')$ ($\delta(\mathbf{x}')$ is the Dirac delta function)
 141 in the direction \mathbf{n} , the displacement variation can be expressed as,

$$142 u_i(\mathbf{x}') = G_{ij}(\mathbf{x}, \mathbf{x}') f_j(\mathbf{x}') \quad (2)$$

143 Due to the position of image terms, the fundamental solution differs whether source point \mathbf{x}' and
 144 field point \mathbf{x} are in the same material phase.

$$145 G_{ij}^y(\mathbf{x}, \mathbf{x}') = \frac{1}{4\pi\mu^w} \begin{cases} \left(\delta_{ij}\phi - \frac{\psi_{,ij}}{4(1-\nu^w)} \right) + A^y \bar{\phi} \delta_{ij} + \chi B^y (\delta_{i3}\delta_{jk} - \delta_{ik}\delta_{j3}) \bar{\alpha}_{,k}^y \\ - C^y x_3 [Q_J \bar{\psi}_{,ij3} + 4(1-\nu^w) \delta_{j3} \bar{\phi}_{,i} + 2(1-2\nu^w) \delta_{i3} Q_J \bar{\phi}_{,j} - Q_J x_3 \bar{\phi}_{,ij}] & x'_3 x_3 \geq 0 \\ - D^y Q_I Q_J \bar{\psi}_{,ji} - (G^y + B^y) Q_I \bar{\beta}_{,ij}^y \\ \left(\delta_{ij}\phi - \frac{\psi_{,ij}}{4(1-\nu^w)} \right) + A^y \phi \delta_{ij} + \chi B^y (\delta_{i3}\delta_{jk} - \delta_{ik}\delta_{j3}) \alpha_{,k}^y \\ - D^y \psi_{,ij} - \chi x_3 F^y \alpha_{,ij}^y - (G^y + B^y) Q_I \beta_{,ji}^y & x'_3 x_3 < 0 \end{cases} \quad (3)$$

146 where the coefficient $\chi = 1$, and superscripts $w ='$ and $y = u$ when $x'_3 \geq 0$; and $\chi = -1$, $w =''$
 147 and $y = l$ when $x'_3 < 0$; $\mathbf{Q} = (1, 1, -1)$ is created for image terms through the interface S and the

149 dummy index rule does not apply to capitalized ones; $\psi = |\mathbf{x} - \mathbf{x}'|$ is the Garlerkin distance vector
 150 and $\phi = \psi^{-1}$; $\frac{\delta_{ij}\phi}{4\pi\mu^w} - \frac{\psi_{,ij}}{16\pi(1-\nu^w)}$ is the Kelvin's solution of infinite space; $\overline{(\cdot)}$ stands for image source
 151 points, such that $\overline{\psi} = |\mathbf{x} - \overline{\mathbf{x}'}|$ and $\overline{\mathbf{x}'} = (x'_1, x'_2, -x'_3)$; $A^u - G^u$ are material constants related to the
 152 upper phase \mathcal{D}^+ ,

$$\begin{aligned}
 A^u &= \frac{\mu' - \mu''}{\mu' + \mu''}, & B^u &= \frac{2\mu'(1 - 2\nu')(\mu' - \mu'')}{(\mu' + \mu'')(\mu' + \mu''(3 - 4\nu'))} \\
 C^u &= \frac{\mu' - \mu''}{2(1 - \nu')(\mu' + (3 - 4\nu')\mu'')}, & D^u &= \frac{3 - 4\nu'}{2}C \\
 F^u &= \frac{2\mu'(\mu'(1 - 2\nu'') - \mu''(1 - 2\nu'))}{(\mu' + \mu''(3 - 4\nu'))(\mu'' + \mu'(3 - 4\nu''))} \\
 G^u &= \frac{\mu'(\mu''(1 - 2\nu'')(3 - 4\nu') - \mu'(1 - 2\nu')(3 - 4\nu''))}{(\mu' + \mu''(3 - 4\nu'))(\mu'' + \mu'(3 - 4\nu''))}
 \end{aligned} \tag{4}$$

154 Similarly, coefficients $A^l - G^l$ can be obtained by switching two material phases, i.e. $A^l = \frac{\mu'' - \mu'}{\mu'' + \mu'}$.

155 The other components in fundamental solution are listed below,

$$\begin{aligned}
 \alpha^u &= \ln[x'_3 - x_3 + \psi], & \overline{\alpha}^u &= \ln[x'_3 + x_3 + \overline{\psi}] \\
 \beta^u &= (x'_3 - x_3)\alpha^u - \psi, & \overline{\beta}^u &= (x'_3 + x_3)\overline{\alpha}^u - \overline{\psi} \\
 \alpha^l &= \ln[-x'_3 + x_3 + \psi], & \overline{\alpha}^l &= \ln[-x'_3 - x_3 + \overline{\psi}] \\
 \beta^l &= (-x'_3 + x_3)\alpha^l - \psi, & \overline{\beta}^l &= (-x'_3 - x_3)\overline{\alpha}^l - \overline{\psi}
 \end{aligned} \tag{5}$$

157 The α functions are also known as Bousinesq's displacement potentials, which are elaborated in
 158 Section "Domain Integral". According to Eqs.(2-4), when $C' = C''$, the coefficients $A - G$ vanishes
 159 and the fundamental solution reduce to Kelvin's solution; When one material phase exhibits zero
 160 stiffness, the fundamental solution reduces to the Mindlin's problem (Yin et al. 2022a).

161 One Inclusion in Two-jointed Dissimilar Half-spaces

162 Consider an infinite domain \mathcal{D} composed of two-jointed dissimilar half-spaces, and one sub-
 163 domain Ω_I is subjected to eigenstrain $\varepsilon_{ij}^*(\mathbf{x})$. Notice that although in Eshelby's work (Eshelby
 164 1957; Eshelby 1959) the eigenstrain is constant over the subdomain, the eigenstrain indeed can
 165 vary spatially. Mura (Mura 1987) found that under interactions of inhomogeneities, the eigenstrain

166 is not uniform anymore, and thus proposed to use the Taylor series expansion at the center of the
 167 subdomain to approximate eigenstrain as Eq.(6),

$$168 \quad \varepsilon_{ij}^*(\mathbf{x}) = \varepsilon_{ij}^{I0*} + (x_p - x_p^{Ic})\varepsilon_{ijp}^{I1*} + (x_p - x_p^{Ic})(x_q - x_q^{Ic})\varepsilon_{ijpq}^{I2*} \quad (6)$$

where \mathbf{x}^{Ic} is centroid of Ω_I subdomain; $\varepsilon^{I0*}, \varepsilon^{I1*}$ and ε^{I2*} are uniform, linear and quadratic components of polynomial to approximate the eigenstrain. The disturbance displacement and strain field caused by eigenstrain can be obtained through the technique of Green's function as follows:

$$u_i(\mathbf{x}) = \int_{\Omega_I} \frac{\partial G_{ij}(\mathbf{x}, \mathbf{x}')}{\partial x'_m} \varepsilon_{kl}^*(\mathbf{x}') C_{jmkl}(\mathbf{x}') dV(\mathbf{x}') = g_{ikl} \varepsilon_{kl}^{I0*} + g_{iklp} \varepsilon_{klp}^{I1*} + g_{iklpq} \varepsilon_{klpq}^{I2*} \quad (7)$$

$$169 \quad 170 \quad \varepsilon_{ij}(\mathbf{x}) = S_{ijkl} \varepsilon_{kl}^{I0*} + S_{ijklp} \varepsilon_{klp}^{I1*} + S_{ijklpq} \varepsilon_{klpq}^{I2*} \quad (8)$$

171 where $g_{iklpq\dots} = \int_{\Omega_I} G_{ij,m'} C_{jmkl} (x'_p - x_p^{Ic})(x'_q - x_q^{Ic}) dV(\mathbf{x}')$ is Eshelby's tensor for displacement;
 172 $S_{pq..ijkl} = \frac{g_{iklpq\dots,j} + g_{jklpq\dots,i}}{2}$ is Eshelby's tensor for strain; indices p, q mean that polynomial-
 173 form terms are involved, i.e. $\psi_p = (x'_p - x_p^{Ic})\psi$. Notice that comparing with Kelvin's solution,
 174 $G_{ij,m'} = -G_{ij,m}$ does not hold for bi-material fundamental solution and the partial derivatives are
 175 provided in Appendix I.

176 Domain Integrals of Fundamental Solution with Polynomial Terms

177 In Eq.(3), the fundamental solution is obtained through superposition of Kelvin's solution
 178 and image terms. As for Kelvin's solution, let Φ and Ψ denote domain integrals of ϕ and ψ ,
 179 respectively. In 1891, Dyson (Dyson 1891) derived the general form domain integrals of elliptical
 180 shell with various density functions. Later Moschovidis and Mura (Moschovidis and Mura 1975)
 181 summarized Dyson's work and defined I and V functions to derive harmonic $\Phi_{pq\dots}$ and biharmonic
 182 $\Psi_{pq\dots}$ potentials, which will not be repeated below.

183 In terms of the image parts, let Θ and Λ denote the domain integrals of α and β , respectively.
 184 Walpole (Walpole 1997) firstly present Θ and Λ based on its definition of Bousinesq's displacement

185 potential and Liu et al. (Liu et al. 2015) extended them up to quadratic order ($\Theta_{pq}, \Lambda_{pq}$). However,
 186 the authors only considered cases of $x'_3 > 0$, thus in the following, we shall complete all cases of
 187 domain integrals with more simplified definitions.

188 In Eq.(5), four types of α and β functions are defined and the use of them is up to locations
 189 of source and field points. Following definition of displacement potentials, the functions can be
 190 rewritten as,

$$191 \quad \begin{aligned} \alpha^u &= \int_{x_3}^{-\infty} \phi(x_1, x_2, t) dt & \bar{\alpha}^u &= \int_{\infty}^{x_3} \bar{\phi}(x_1, x_2, t) dt \\ \alpha^l &= \int_{\infty}^{x_3} \phi(x_1, x_2, t) dt & \bar{\alpha}^l &= \int_{x_3}^{-\infty} \bar{\phi}(x_1, x_2, t) dt \end{aligned} \quad (9)$$

192 where, only finite part of Eq.(9) are considered since infinite constant vanishes during partial
 193 differentiation. Similarly, β functions can be written as Eq.(9) with the same integral limits but
 194 switch integral functions from ϕ or $\bar{\phi}$ to α or $\bar{\alpha}$, respectively. Because both α and β functions are
 195 defined through integrals along the third axis with respect to field point, one can interchange the
 196 sequence of integral, taking Θ^u as an example,

$$197 \quad \begin{aligned} \Theta^u &= \int_{\Omega_I} \alpha^u dV(\mathbf{x}') = \int_{x_3}^{-\infty} \int_{\Omega_I} \phi dV(\mathbf{x}') dt = \int_{x_3}^{-\infty} \Phi(x_1, x_2, t) dt \\ \Lambda^u &= \int_{\Omega_I} \beta^u dV(\mathbf{x}') = \int_{x_3}^{-\infty} \int_{\Omega_I} \alpha^u dV(\mathbf{x}') dt = \int_{x_3}^{-\infty} \Theta^u(x_1, x_2, t) dt \end{aligned} \quad (10)$$

198 Notice that for the integral with respect to t along the third axis does not include any integral
 199 points inside the subdomain, hence, only the exterior branch of Φ and Θ^u is retained. Following the
 200 same fashion, other Θ, Λ and their polynomial involved functions can be derived. In the following,
 201 their integrals are provided as below. The superscript s represents 4 types of functions defined in
 202 Eq.(9), which is up to locations of source and field points. $\phi^s = \bar{\phi}, \psi^s = \bar{\psi}$ when $x'_3 x_3 \geq 0$ and
 203 $\phi^s = \phi, \psi^s = \psi$ when $x'_3 x_3 < 0$

204 *Uniform Domain Integrals Θ and Λ*

$$205 \quad \Theta^s = \frac{4\pi a^3}{3} \alpha^s \quad \& \quad \Lambda^s = \frac{4\pi a^3}{3} \beta^s \quad (11)$$

206 *Linear Domain Integrals Θ_p and Λ_p*

$$207 \quad \Theta_p^s = \frac{4\pi a^5}{15} \begin{cases} -\alpha_{,p}^s & p \neq 3 \\ \phi^s & p = 3 \end{cases} \quad \& \quad \Lambda_p^s = \frac{4\pi a^5}{15} \begin{cases} -\beta_{,p}^s & p \neq 3 \\ \alpha^s & p = 3 \end{cases} \quad (12)$$

208 *Quadratic Domain Integrals Θ_{pq} and Λ_{pq}*

$$209 \quad \Theta_{pq}^s = \frac{4\pi a^5}{105} \begin{cases} -a^2 \alpha_{,q}^s (\alpha_{,p}^s + \ln[\psi^s]_{,p}) + \delta_{pq} [7\alpha^s + a^2 \phi^s \gamma^s] & p, q \neq 3 \\ -a^2 \phi_{,p}^s & p \neq 3, q = 3 \\ a^2 \phi_{,3}^s + 7\alpha^s & p = q = 3 \end{cases} \quad (13)$$

210 and

$$211 \quad \Lambda_{pq}^s = \frac{4\pi a^5}{105} \begin{cases} -a^2 (x_p - x_p^{Ic}) \gamma_{,q}^s + \delta_{pq} [7\beta^s - a^2 \gamma^s] & p, q \neq 3 \\ -a^2 \alpha_{,p}^s & p \neq 3, q = 3 \\ a^2 \alpha_{,3}^s + 7\beta^s & p = q = 3 \end{cases} \quad (14)$$

212 where γ^s is argument of the logarithmic function (α^s). When $\alpha^s = \alpha^u$, we can obtain $\gamma^s = x'_3 - x_3 + \psi$.

213 **One Inhomogeneity in a Bounded Bi-material Domain**

214 In the last subsection, the disturbance from polynomial-form eigenstrain can be obtained through
 215 the explicit domain integral of fundamental solution over the spherical subdomain Ω_I . Combining
 216 with the conventional BEM, the elastic field is superposition of boundary responses with BIEs and
 217 disturbance of eigenstrain with Eq.(7) and Eq.(8), and the displacement of arbitrary field point \mathbf{x}
 218 within \mathcal{D}^+ can be expressed as,

$$219 \quad u_i(\mathbf{x}) = - \int_{\partial\mathcal{D}^t} T_{ij}(\mathbf{x}, \mathbf{x}') u_j(\mathbf{x}') d\mathbf{x}' + \int_{\partial\mathcal{D}^u} G_{ij}(\mathbf{x}, \mathbf{x}') t_j(\mathbf{x}') d\mathbf{x}' + \int_{\Omega_I} \frac{\partial G_{ij}(\mathbf{x}, \mathbf{x}')}{\partial x'_m} \varepsilon_{kl}^*(\mathbf{x}') C_{jmkl}(\mathbf{x}') dV(\mathbf{x}') \\ = - \sum_{e=1}^{NE} H_{ij} u_j^e + \sum_{e=1}^{NE} U_{ij} t_j^e + g_{ikl} \varepsilon_{kl}^{I0*} + g_{iklp} \varepsilon_{klp}^{I1*} + g_{iklpq} \varepsilon_{klpq}^{I2*} \quad (15)$$

220 where G_{ij} and $T_{ij} = \frac{C_{imkl}(\mathbf{x}') (G_{kj,l'} + G_{lj,k'})}{2} n_m(\mathbf{x}')$ are fundamental solution to displacement and
 221 traction, respectively. With boundary surface mesh, the BIEs are expressed in a discretization form
 222 (Beer et al. 2008) that $H_{ij} = \int_{S_e} T_{mi}(\mathbf{x}, \mathbf{x}') N_{mj}(\mathbf{x}') dS$ and $U_{ij} = \int_{S_e} U_{mi}(\mathbf{x}, \mathbf{x}') N_{mj}(\mathbf{x}') dS$; NE is
 223 the number of elements; the superscript e represents nodal values of boundary displacements and
 224 surface tractions in the e^{th} element. In Eq.(15), the interactions between subdomains and boundary
 225 are involved to displacement of BEM. Since the continuity conditions of bi-material interface S
 226 has been analytically considered in the fundamental solutions, one can solve the boundary response
 227 similar to a homogeneous matrix.

228 Unlike inclusion problems with prescribed eigenstrain, when the Ω^I is filled with different
 229 material C^I , the eigenstrain is yet to be determined with equivalent stress conditions. Mura (Mura
 230 1987) proposed the conditions for polynomial-form eigenstrain to simulate material mismatch,

$$\begin{aligned}
 C_{ijkl}^w(\varepsilon_{kl}^b + \varepsilon'_{kl} - \varepsilon_{kl}^{I0*}) &= C_{ijkl}^I(\varepsilon_{kl}^b + \varepsilon'_{kl}) \\
 C_{ijkl}^w(\varepsilon_{kl,m}^b + \varepsilon'_{kl,m} - \varepsilon_{klm}^{I1*}) &= C_{ijkl}^I(\varepsilon_{kl,m}^b + \varepsilon'_{kl,m}) \\
 \frac{1}{2!} C_{ijkl}^w(\varepsilon_{kl,mn}^b + \varepsilon'_{kl,mn} - 2\varepsilon_{klmn}^{I2*}) &= \frac{1}{2!} C_{ijkl}^I(\varepsilon_{kl,mn}^b + \varepsilon'_{kl,mn})
 \end{aligned} \tag{16}$$

232 Because the inhomogeneity may be located in either the upper phase \mathcal{D}^+ or the lower phase \mathcal{D}^- , in
 233 Eq.(16), the superscript $w ='$ when Ω^I is located is located in \mathcal{D}^+ and $w =''$ when Ω^I is located is
 234 located in \mathcal{D}^- . ε_{ij}^b is strain contributed by BIEs of boundary response with Eq.(20) in Appendix
 235 I; ε'_{ij} is disturbed strain field expressed in Eq.(8). As indicated in Eq.(16), the interaction of
 236 inhomogeneities are taken into account in equivalent stress conditions that one inhomogeneity can
 237 disturb stress field of another inhomogeneity. Therefore, assembling the conventional BEM matrix,
 238 collecting eigenstrain effects on boundary nodes and the stress equivalent equations, the iBEM
 239 global system of linear equations will be shown in Eq. (19) of Appendix I.

240

241 Discussion and extension to ellipsoidal inhomogeneity

242 In this article, the above single-domain iBEM algorithm aims to save efforts in handling of trivial

procedures for domain discretization of the inhomogeneities. As for the industrial applications, Koenigsberger et al. (Königsberger et al. 2020) proposed a novel scheme on combination of Dvorak's transformation field analysis and Eshelby's method to investigate poro-elastic properties of cement paste, where the ellipsoidal Eshelby's tensor is introduced relating eigenstresses. In (Buchner et al. 2021) demonstrated experimental investigation on the effective elastic and thermal properties of clay bricks and various shapes of defects, i.e mesopores, quartz, are considered. Since the algorithm is based on the bi-material Green's function, and particularly the polynomial-form Eshelby's tensor used in equivalent stress conditions. Therefore, the limitations of the algorithm is the same limitation in the domain integral of the Green's function. Shown in Section "Domain integrals of fundamental solution with polynomial terms", the components of spherical Eshelby's tensor are derived by interchanging the integral sequence. Following the same fashion, (i) the domain integrals Φ and Ψ over an ellipsoidal region can be found in (Dyson 1891) and (Moschovidis and Mura 1975); (ii) the domain integrals Θ and Λ can be derived by partial integration along the third axis given integral limits. Notice that although Mura (Mura 1987) proposed the Taylor series expansion of eigenstrain to handle interactions of ellipsoidal inhomogeneities, when the ratios of axes of ellipsoid become too large / small, even quadratic eigenstrain terms may not provide accurate solution due to large variations of eigenstrain.

260 **NUMERICAL VERIFICATION**

261 The aforementioned algorithm is implemented to software package of iBEM to predict local
262 fields of composites with prescribed boundary conditions. In order to validate the algorithm and
263 illustrate how solutions with 3 orders of polynomial eigenstrain performs, a numerical case study
264 with Robin's boundary condition is set up. Eshelby's solution of an inhomogeneity embedded in
265 infinite space provides insight of uniform eigenstrain distribution. Actually, for a two-jointed half-
266 spaces, if the inhomogeneity is far from the bi-material interface S , the image terms in fundamental
267 solution Eq.(3) and its domain integral vanishes rapidly, which provides minor effects on stress
268 disturbance. As a consequence, the elastic fields become similar to Eshelby's solution of infinite
269 space. Therefore, the inhomogeneity will be placed close to the interface S to observe interfacial

270 effects on elastic fields. For spherical inhomogeneities, the intensity of interaction is commonly
271 judged by the ratio $d = h/a$, where a is radius and h is distance from the centroid to interface.
272 Following the definition of Fig.1, the dimensions and boundary conditions are set as: (i) the width
273 b , length l and two thickness T_1, T_2 are set as 1 m; (ii) the inhomogeneity with radius $a = 0.1$ m is
274 placed in \mathcal{D}^- , where the distances h to interface S are $h = 1.1a, 2a$ and $3a$; (iii) shown in Fig.2,
275 the top surface of \mathcal{D}^+ is subjected uniform downward pressure 1 MPa; and the displacement of
276 bottom surface of \mathcal{D}^- is constrained; all other four surfaces are free of traction. The shear modulus
277 and Poisson's ratio for two material are $\mu' = 0.4$ MPa, $\nu' = 0.25$ and $\mu'' = 0.8$ MPa and $\nu'' = 0.1$,
278 respectively.

279 Shown in Fig.3(a) and Fig.3(b), the elastic fields u_3 and σ_{33} are compared with FEM with
280 different ratio of distance. Because the disturbance by the inhomogeneity vanishes rapidly, the
281 $[-3, 3]a$ around the inhomogeneity is considered. In simulation, 1000 quadrilateral boundary
282 elements are used in iBEM and 2, 189, 461 tetrahedral elements are used in FEM. Regarding u_3 ,
283 the curves agree well with FEM except minor discrepancy is observed in the second branch of
284 case $h/a = 1.1$. In terms of stress comparison, the main discrepancy between iBEM and FEM
285 exists at the interface between inhomogeneity Ω_I and matrix, which can be interpreted as intensive
286 interfacial effects. Indicated in Fig.3(c), the stress concentration factors are compared in the vertical
287 hoop of sphere and the maximum $\sigma_{\theta\theta}$ changes with intensivity of interfacial effects, specifically, in
288 the case $h/a = 1.1$, the angle shifts near 14 degree. As the h/a increases, both stress concentration
289 and angle shift decreases, which will be elaborated in "Ratio of Distance". In Fig.3(d), the stress
290 σ_{33} ($h/a = 1.2$) is compared along the center line among uniform, linear and quadratic series
291 expansion. It is observed that the assumption of uniform distribution (Eshelby's solution) cannot
292 predict the variation of elastic fields. The linear terms improves the accuracy but exhibit obvious
293 discrepancy in the neighborhood of the inhomogeneity. Therefore, the introduction to quadratic
294 term is necessary in improvement of accuracy of solution. **In the comparison among uniform, linear**
295 **and quadratic order terms, as shown in Fig. 3(d), the accuracy of predictions improves with the**
296 **increase of polynomial order. Such phenomenon indicates that due to the existence of interfacial**

297 effects, none of the numerical solution (uniform, linear and quadratic) is exact solution and so does
298 the FEM. Back to Fig. 3(c), except the curve $h/a = 1.1$, the other two cases $h/a = 2$ and $h/a = 3$
299 agree well with FEM. When the interfacial effects dominate, much more disturbance is observed,
300 therefore, the disturbance also make the eigenstrain field more complex, which is the reason why
301 even quadratic eigenstrains cannot provide adequately accuracy as iBEM exhibits an angle lag.
302 When the distance ratio h/a increases and interfacial effects decrease accordingly, the uniform
303 term gradually dominates the solution. In such case, only uniform term alone can provide accurate
304 results (Wu and Yin 2021). Although the iBEM algorithm has enabled three tailororable accuracy
305 options, in the following, quadratic term is applied to ensure reliable and accurate analysis.

306 CASE STUDY OF STRESS CONCENTRATION FACTORS

307 In the previous section, the iBEM algorithm has been validated with FEM with three cases of
308 different distance ratios in a bi-material system. In this section, the algorithm is further utilized to
309 investigate elastic fields and stress concentration factors of a specific industrial application with a
310 microvoid $a = 2.5 \times 10^{-6}$ m in the sunslate. Because of the thin adhesive layer, the sunslate can be
311 considered as a bi-material system with $l = b = 0.1$ m of (i) glass ($E' = 72$ GPa, $\nu' = 0.2$, T_1) and
312 (ii) concrete panels ($E'' = 36$ GPa, $\nu'' = 0.2$, T_2) (Yin et al. 2022b), as shown in Fig.1. The sunslate
313 is subjected to downward pressure and thermal loads, which can be decomposed of superposition
314 of free expansion and applied pressure for mismatch of thermal expansion coefficient (Yin and
315 Prieto-Muñoz 2013), as shown in Fig.2. Since the free expansion does not result in variation of
316 stress, this paper focus the second parts and the applied loads are set as 1 MPa, which can be easily
317 extended for other values of loads in linear elastic stage.

318 Ratio of Distance

319 The interfacial effects are judged through ratio of distance between the inhomogeneity and inter-
320 face S . In this subsection, 8 ratio of distances are considered as $h/a = 1.1, 1.2, 1.4, 1.6, 1.8, 2, 2.5, 3$
321 and the microvoid is placed in lower and upper phase for overall 16 cases. In Fig.4(a) and Fig.4(b),
322 the stresses $\sigma_{\theta\theta}$ are plotted versus the vertical hoop angle $\theta \in [-90^\circ, 90^\circ]$ when the microvoid is
323 located in upper and lower phase, respectively. Comparing the angle of maximum stress, when

324 h/a increases and interfacial effects decrease, the angle gradually decreases accordingly. If h/a is
 325 large enough, the contribution from image terms in fundamental solutions vanishes and angle of
 326 maximum stress changes as zero, which is similar to a infinite space problem but with disturbed
 327 elastic fields due to the existence of the other phase. Obviously, the material mismatch influence
 328 the stress concentration as well. Indicated in Fig.4(a) and Fig.4(b), when microvoid is embedded
 329 in \mathcal{D}^+ , the change of distance ratio exhibits larger variation of both angle shifts and stress concen-
 330 tration factor. In Fig.5(a), the angle shift in \mathcal{D}^+ is over 3 times than that of \mathcal{D}^- , which explains
 331 larger material mismatch amplify interfacial effects. Fig.4(c) shows the displacement u_3 of 6 cases
 332 of microvoid under uniform downward pressure 1 MPa on the vertical hoop. When the distance
 333 between microvoid and bottom surface decreases, the displacement u_3 decreases accordingly due
 334 to constraint of bottom surface. When angle $\theta = -90^\circ$ and microvoid is embedded in \mathcal{D}^+ , the
 335 differences between case "h/a = 1.1 - Upper" and "h/a = 1.2 - Upper" is larger than other angles.
 336 Such phenomenon is caused by interfacial effects associated with softer material concrete in \mathcal{D}^- .
 337 Similarly, when angle $\theta = 90^\circ$ and microvoid is embedded in \mathcal{D}^- , opposite trend is observed.
 338 Shown in Fig.4(d), under horizontal pressure, the distance ratio has minor effects on stress concen-
 339 tration and angle shifts, which can be interpreted as the direction of load is perpendicular to x_3 .
 340 Fig.5(b) exhibits the variation of stress concentration factor versus h/a with two types of loads and
 341 positions of microvoid. When h/a increases, the stress concentration factors tends to be constant,
 342 which is similar to Kelvin's problem.

343 **Ratio of Shear Moduli**

344 This subsection aims to present how ratio of shear moduli influence the stress concentration
 345 caused by the microvoid. Discussed in "Ratio of distance" section, besides the interfacial effects,
 346 the material mismatch also contribute to stress concentration. To avoid such factors, the microvoid
 347 is placed in the upper phase \mathcal{D}^+ with distance ratio $h/a = 1.2$ and the material properties of the
 348 upper phase remain the same ($E' = 72\text{GPa}$, $\nu' = 0.2$). 9 ratios of shear moduli $\frac{\mu'}{\mu''}$ are considered as
 349 0.05, 0.1, 0.2, 0.5, 1, 2, 5, 10 and 20. Shown in Fig.6(a) and Fig.6(b), the stress concentration factors
 350 increases with μ'/μ'' and angle of maximum hoop stress shifts to negative range. When μ'/μ''

351 is small, the stiffer lower phase restrict the displacements and vice versa; consequently, as partial
352 derivatives of displacements, the stress concentration factor is smaller and angle of maximum stress
353 shifts to positive range. In Fig.6(c), when $\mu'/\mu'' = 1$, the fundamental solution reduces to Kelvin's
354 solution, the stress concentration factor under vertical load is close to 2. Fig.6(d) plots the angle of
355 maximum stress versus ratios of shear moduli and vertical load case has larger variations because
356 the loading direction is perpendicular to the interface S . The angles of maximum stress in two
357 curves are close to 5° , which is caused by non-uniformity of strain field around the microvoid (not
358 a far-field uniform strain).

359 **Ratio of Thickness**

360 This subsection aims to investigate thickness ratio effects on the stress concentration behavior
361 caused by the microvoid. Following the "Ratio of shear moduli", the microvoid is placed in \mathcal{D}^+ with
362 distance ratio $h/a = 1.2$ and 6 ratios 1, 2, 5, 10, 15 and 20 are selected. Fig.7(a) plots the vertical
363 hoop stress with hoop angle. It is noticed that when $T_1/T_2 \leq 10$ and the ratio increases, the negative
364 stress concentration factor decreases and positive stress concentration factor increases obviously.
365 However, from case 15 and case 20, the trend reverses that case 20 has similar stress distribution as
366 case 5; case 15 exists between case 10 and 5. Such phenomenon is caused by the elastic behavior of
367 thin panels or plates. When the thickness ratio $T_1/T_2 \leq 10$, under horizontal pressure, the primary
368 forces for \mathcal{D}^+ are bending and shear; as thickness ratio exceeds 10, the lateral shearing deformation
369 become neglecting, which explains the maximum negative stress concentration increases due to
370 elastic behavior change. Fig.7(b) indicates the trend of stress concentration factor discussed above.
371 Under horizontal pressure, the horizontal hoop stress has similar trends as the other two curves.
372 Notice that, unlike distance ratio and shear moduli ratio, the thickness ratio seldom change the
373 angle of maximum negative/positive hoop stress. Because the angle shifts of maximum hoop stress
374 is generally considered as interfacial effects, the smaller distance ratio, larger material mismatch
375 and shear moduli ratio can augment such effects. However, the change of stress distribution with
376 thickness ratios is mainly caused by the elastic behavior of the bi-material system itself.

377 **CONCLUSIONS**

378 The algorithm of the single domain inclusion-based boundary element method has been applied
379 to investigate the elastic fields of bi-material system embedded with one microvoid. The algorithm
380 has been verified by FEM with case study of void embedded in a two-cuboid bi-material system
381 with different distance ratios. Along with the numerical verification, the iBEM with uniform, linear
382 and quadratic terms are compared with FEM and provide tailorble accuracy upon readers' needs.
383 Thanks to the fundamental solution of bi-material, and explicit domain integrals, the conventional
384 boundary value problem with interface and subdomain can be solved similarly to a homogeneous
385 solid. In this paper, the algorithm has been applied to study the stress concentration issues arisen
386 in a Sunslate and parametric studies are conducted on effects brought by distance ratio, shear
387 modulus ratio and thickness ratio and loading conditions. The algorithm is particularly suitable for
388 understanding, designing and conducting virtual experiments on a thin-film system with potential
389 defects.

390 **Data Availability Statement**

391 All data, models, or code that support the findings of this study are available from the corre-
392 sponding author upon reasonable request.

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397 **APPENDIX I.**

398 **Method of images in Walpole's solution and its modified form**

399 This appendix subsection aims to provide details and discussion of fundamental solution by
400 Walpole (Walpole 1996). For a perfect bounded bi-material interface, the continuity conditions in
401 Eq. (1) on displacements and tractions need to be satisfied. The method of images is a typical
402 mathematical tool to solve partial differential equations that a mirror image source is artificially
403 created to handle the continuity conditions of the interface without extending the domain of function.
404 Haberman (Haberman 2021) illustrated the method through semi-infinite solution to Poisson's
405 equation in Chapter 9.5.8 and transient heat transfer in Chapter 11.5.3. Walpole (Walpole 1996)
406 wrote the fundamental solution (displacement) in terms of two branches, which are determined by
407 positions of field and source points. Subsequently, similar to Boussinesq's solution and Mindlin's
408 problem, the potential functions are assumed as partial integration with respect to the third axis,
409 where part of Kelvin's solution and nuclei of strain are applied. According to the continuity
410 conditions, the coefficients associated with material constants can be determined.

411 The original form of bi-material elastic fundamental solution can be applied to investigated induced
412 elastic fields caused by loads at any arbitrary interior point. However, its application to Eshelby's
413 problem is not complicated and trivial as Eshelby's tensor leads domain integrals with the free
414 source terms, i.e x'_3 . It is possible to represent the free source terms with Garlekin's distance vector,
415 for example, the domain integral of $x'_3\phi$ can altered as $\int_{\Omega}(x'_3 - x_3) + x_3\phi d\mathbf{x}' = -\Psi_{,3} + x_3\Phi$, which
416 simplifies the domain integral expressions.

417 **Partial Derivatives of Domain Integrals**

418 It is noted that with image terms, partial differentiation process changes accordingly, say
419 $\bar{\alpha}_{,i'}^u = -Q_I \bar{\alpha}^u$, we provide the first order derivative of fundamental solution to obtain Eshelby's
420 tensor for displacement with quadratic order polynomial,

421 (1) When $x'_3 x_3 \geq 0$,

$$\begin{aligned}
& 4\pi\mu^w \int_{\Omega} (x'_p - x_p^c)(x'_q - x_q^c) G_{ij,m'} dV(\mathbf{x}') \\
&= (-\delta_{ij}\Phi_{pq,m} + \frac{\Psi_{pq,ijm}}{4(1-\nu^w)}) - Q_P Q_Q \left\{ A^y \delta_{ij} Q_M \bar{\Phi}_{pq,m} - \chi B^y Q_M (\delta_{i3} \bar{\Theta}_{pq,jm}^y - \delta_{j3} \bar{\Theta}_{pq,im}^y) \right. \\
&\quad \left. - C^y x_3 Q_M \left[-Q_J \bar{\Psi}_{pq,ij3m} - 4(1-\nu^w) \delta_{j3} \bar{\Phi}_{pq,im} - 2(1-2\nu^w) \delta_{i3} Q_J \bar{\Phi}_{pq,jm} + x_3 Q_J \bar{\Phi}_{pq,im} \right] \right. \\
&\quad \left. + Q_J Q_M \left[D^c Q_I \bar{\Psi}_{pq,ijm} + (G^y + B^y) \bar{\Lambda}_{pq,ijm}^y \right] \right\} \\
&\quad (17)
\end{aligned}$$

423 (2) When $x'_3 x_3 < 0$,

$$\begin{aligned}
& 4\pi\mu^w \int_{\Omega} (x'_p - x_p^c)(x'_q - x_q^c) G_{ij,m'} dV(\mathbf{x}') \\
&= (-\delta_{ij}\Phi_{pq,m} + \frac{\Psi_{pq,ijm}}{4(1-\nu^w)}) - A^y \delta_{ij} \Phi_{pq,m} - \chi B^y (\delta_{i3} \Theta_{pq,jm}^y - \delta_{j3} \Theta_{pq,im}^y) \\
&\quad + D^y \Psi_{pq,ijm} + \chi x_3 F^y \Theta_{pq,ijm}^c + (G^y + B^y) Q_I \Lambda_{pq,ijm}^y
\end{aligned} \quad (18)$$

425 For other higher order derivatives to obtain Eshelby's tensor for strain is straightforward because
426 the partial differentiation is with respect to x_n not x'_n .

427 **Global Matrix of iBEM**

$$\begin{bmatrix}
\mathcal{H} + H & \dots & -g^{0I} & -g^{1I} & -g^{2I} & \dots \\
\vdots & \vdots & \vdots & \vdots & \vdots & \vdots \\
\Delta C^I H^{1I} & \dots & \Delta C^I S^{0I} & \Delta C^I S^{1I} & \Delta C^I S^{2I} & \dots \\
\Delta C^I H^{2I} & \dots & \Delta C^I S^{0I'} & \Delta C^I S^{1I'} & \Delta C^I S^{2I'} & \dots \\
\Delta C^I H^{3I} & \dots & \Delta C^I S^{0I''} & \Delta C^I S^{1I''} & \Delta C^I S^{2I''} & \dots \\
\vdots & \vdots & \vdots & \vdots & \vdots & \vdots
\end{bmatrix}_{(3*NN+s*NI) \times (3*NN+s*NI)} \begin{bmatrix} u \\ \vdots \\ \varepsilon^{*0I} \\ \varepsilon^{*1I} \\ \varepsilon^{*2I} \\ \vdots \end{bmatrix}_{(3*NN+s*NI)} = \begin{bmatrix} U \\ \vdots \\ -\Delta C^I U^{1I} \\ -\Delta C^I U^{2I} \\ -\Delta C^I U^{3I} \\ \vdots \end{bmatrix}_{(3*NN+s*NI) \times (3*NN)} \begin{bmatrix} t \\ \vdots \end{bmatrix}_{(3*NN) \times (3*NN)} \quad (19)$$

428 where s is determined order of polynomial by users, as 6, 24, 60 for uniform, linear and quadratic;
 429 \mathcal{H} is a diagonal matrix applied in conventional BEM to eliminate strong singularities with method
 430 of rigid body motion for static problems; H and U are coefficients calculated by discretized BIE in
 431 Eq.(15); g and S are Eshelby's tensor for displacement and strain (with uniform, linear and quadratic
 432 order), respectively; H^{1I}, H^{2I}, H^{3I} and U^{1I}, U^{2I}, U^{3I} are coefficients calculated by discretized BIE
 433 in Eq.(20) below for strain, first order partial derivative of strain and second order partial derivatives
 434 of strain, respectively. They are used in equivalent stress conditions as Eq.(16).

$$435 \quad \begin{aligned} U_{ijm}^{0I} &= \frac{1}{2} \int_{S_e} (H_{ik,j}(\mathbf{x}, \mathbf{x}') + H_{jk,i}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS \\ H_{ijm}^{0I} &= \frac{1}{2} \int_{S_e} (T_{ik,j}(\mathbf{x}, \mathbf{x}') + T_{jk,i}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS \end{aligned} \quad (20)$$

436 Similarly, other higher order partial derivatives can be obtained, i.e., $U_{ijmnr}^{2I} = \frac{1}{2} \int_{S_e} (H_{ik,jnr}(\mathbf{x}, \mathbf{x}') +$
 437 $H_{jk,inr}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS$.

438

439 **APPENDIX II. COMPUTATIONAL TESTS BETWEEN IBEM AND FEM**

440 This appendix section aims to provide details in computational efficiency and accuracy compar-
441 ison between the proposed iBEM programmed by authors and FEM using a commercial software
442 ANSYS. Regarding the accuracy and convergence, two aspects are discussed, (a) a convergence
443 analysis of FEM through refinement of elements around the inhomogeneity; and (b) an error analy-
444 sis concerning the differences between iBEM and FEM (convergent results). As for the efficiency,
445 three main aspects are considered, (i) preparation stage, such as generation of geometric specifica-
446 tion ,surface (iBEM) / volume (FEM) domain discretization; (ii) construction and solving process,
447 such as computation of global matrix and solution time; and (iii)a comparison occupancy test of
448 CPU and RAM in two methods.

449 Without the loss of any generality, this section follows "Numerical Verification" that all material
450 properties are retained and the ratio of distance $h/a = 1.5$, where the inhomogeneity with radius
451 $a = 0.1$ m is located in the lower phase \mathcal{D}^- . For BEM surface mesh, 4-node bi-linear quadrilateral
452 elements with 4 Gauss integral points are used with adaptive subdivision integration scheme
453 following (Eberwien et al. 2005). Regarding FEM volume mesh, 10-node quadratic tetrahedral
454 elements are used due to expected larger variation of displacements.

455

456 **Accuracy test**

457 Shown as Fig. 8, in order to use fewer elements, the neighbor box (dimension $0.4 \times 0.4 \times 0.3$ m)
458 with the spherical inhomogeneity and the rest region of the matrix are treated with two mesh sizes.
459 Four internal size steps are selected as 0.04, 0.015, 0.008 and 0.006 m, while the uniform external
460 mesh size is 0.04 m. When the internal mesh size decreases, the number of elements increase
461 accordingly as 41, 4906, 50, 9469, 114, 7928 and 232, 6729. Indicated in Fig. 9(a), the stress
462 concentration factor $\sigma_{\theta\theta}/\sigma_{33}^0$ gradually converges as the differences between size steps decrease.
463 Particularly, size step 0.006 and 0.008 exhibit very minor discrepancies, and the two curves also
464 agree well with results in iBEM. Considering the larger differences in Fig. 3(c), when the distance
465 ratio h/a increase, the variation in eigenstrain decreases accordingly, which can provide good

466 predictions. Fig. 9(b) indicates the errors between iBEM and two FEM curves, although there are
467 some differences, considering the maximum stress concentration factor is 1.94, the errors between
468 two methods are acceptable.

469

470 **Efficiency test**

471 Shown as Table. 1, FEM and iBEM-quadratic used 2 and 8 cores, respectively. We keep
472 the default setting of ANSYS, which limits participation of more cores, so that the solving time
473 will be shorter. Since the solving process contains calculation, assignment of coefficients and
474 matrix decomposition, the solution time is not linearly proportional to number of cores, because
475 assignment of coefficients is partially a single-thread process. Although Table. 1 indicates iBEM
476 package occupies all cores of the CPU, it uses "Eigen" library, where the process decomposition
477 of matrix is not fully multi-core. The iBEM-quadratic exhibit apparent advantages on RAM usage
478 over FEM, because the degree of freedom is a constant as 3,060. Regarding the mesh process,
479 iBEM only require the surface mesh, which avoids trivial process on the inhomogeneity and its
480 neighbor region as shown in Fig. 8 (d). In addition, FEM require specification of models, such
481 as importing from AutoCAD or creating in its own geometry editors, which requires more efforts.
482 Considering all above factors, iBEM would be an efficient and computational resource friendly
483 scheme.

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TABLE 1. Comparison of efficiency among iBEM with quadratic terms and FEM with four internal size steps

	CPU Usage (cores)	RAM Usage (Gb)	Mesh time (s)	Solving time (s)
FEM-0.04	2	2.22	17	48
FEM-0.015	2	3.51	27	58
FEM-0.008	2	7.65	39	133
FEM-0.006	2	12.60	72	326
iBEM-quadratic	8	0.58	5	10.2
CPU: i7-9700K (8 cores)	RAM: 2933mHz			

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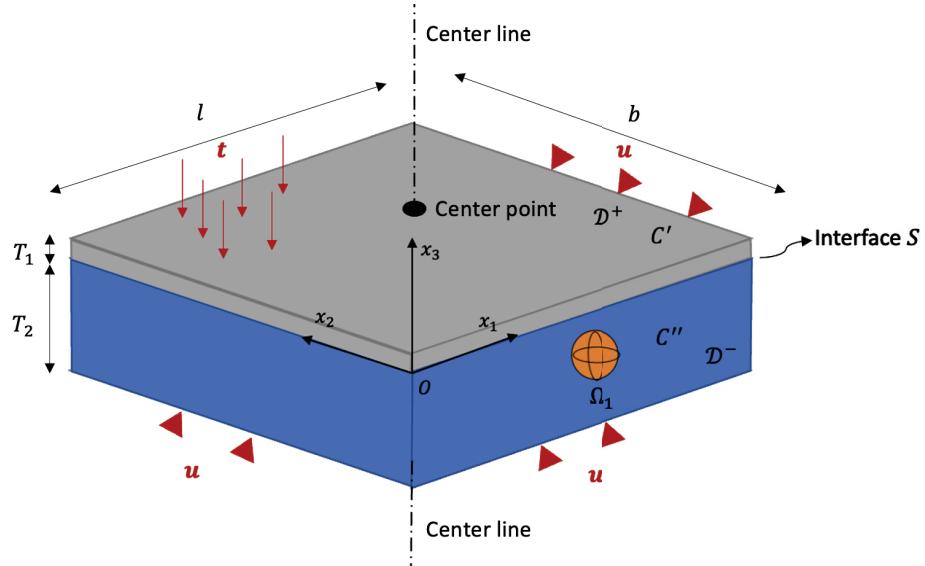


Fig. 1. Schematic plot of a bi-material system \mathcal{D} (a) subjected to mixed prescribed boundary conditions embedded with one inhomogeneity Ω_I

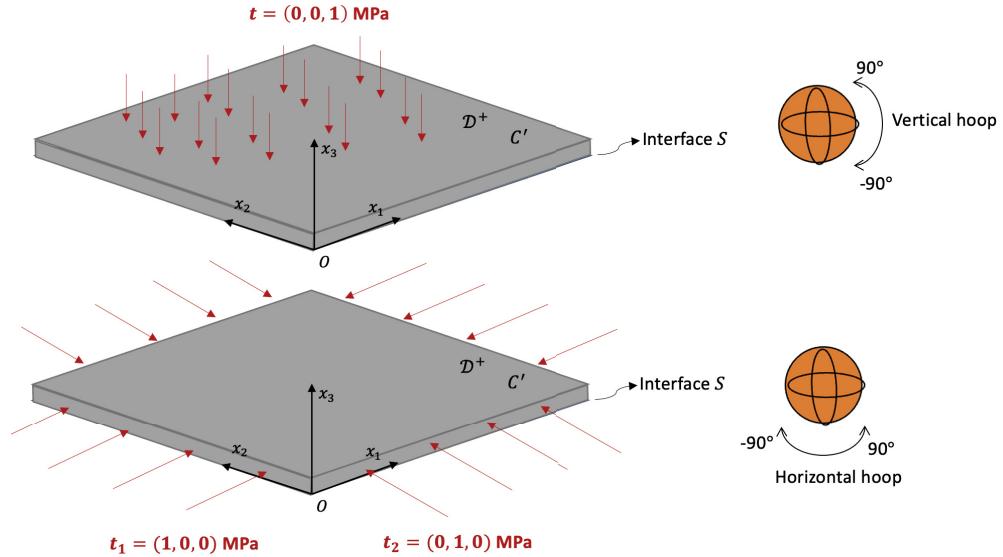


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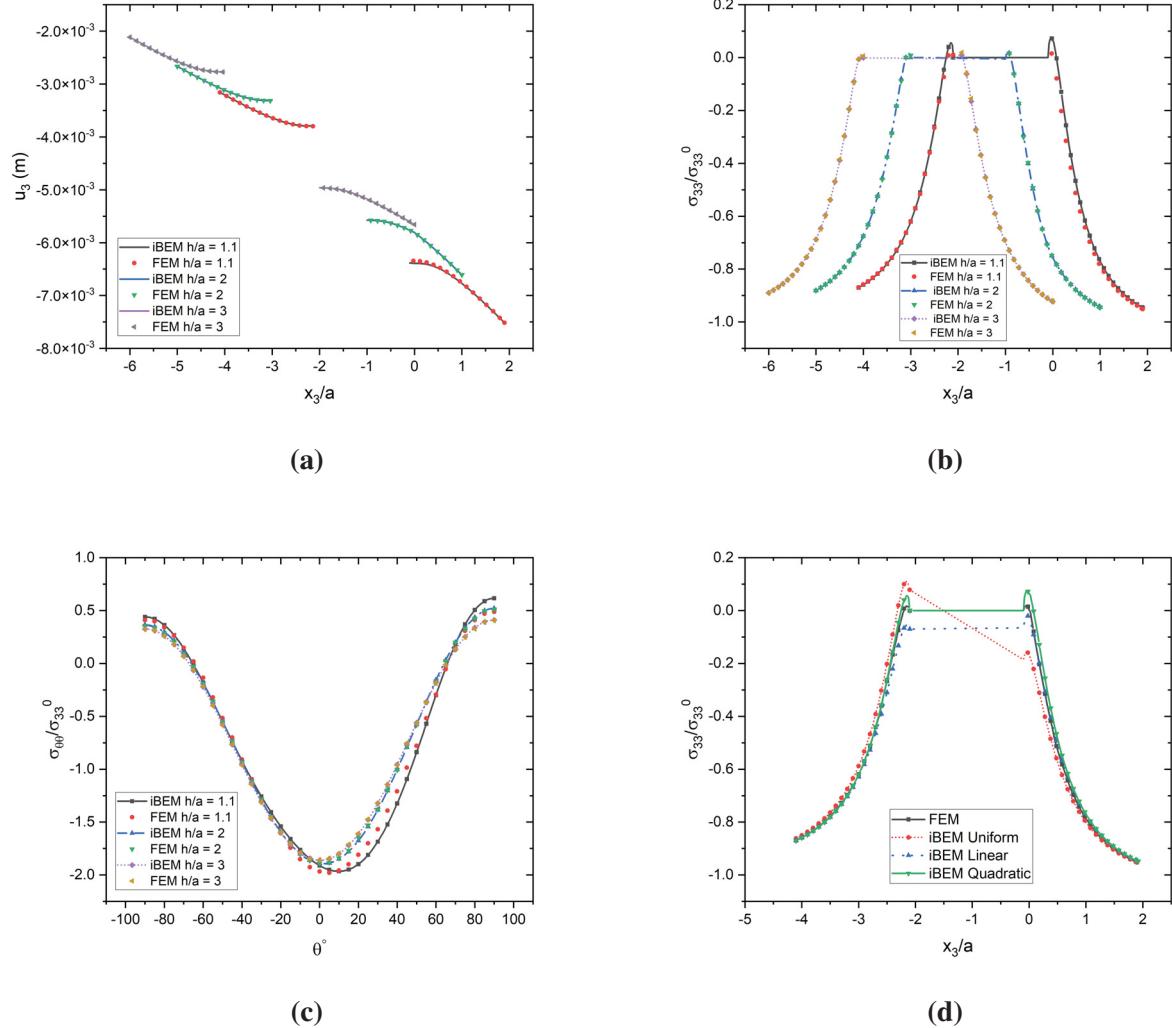


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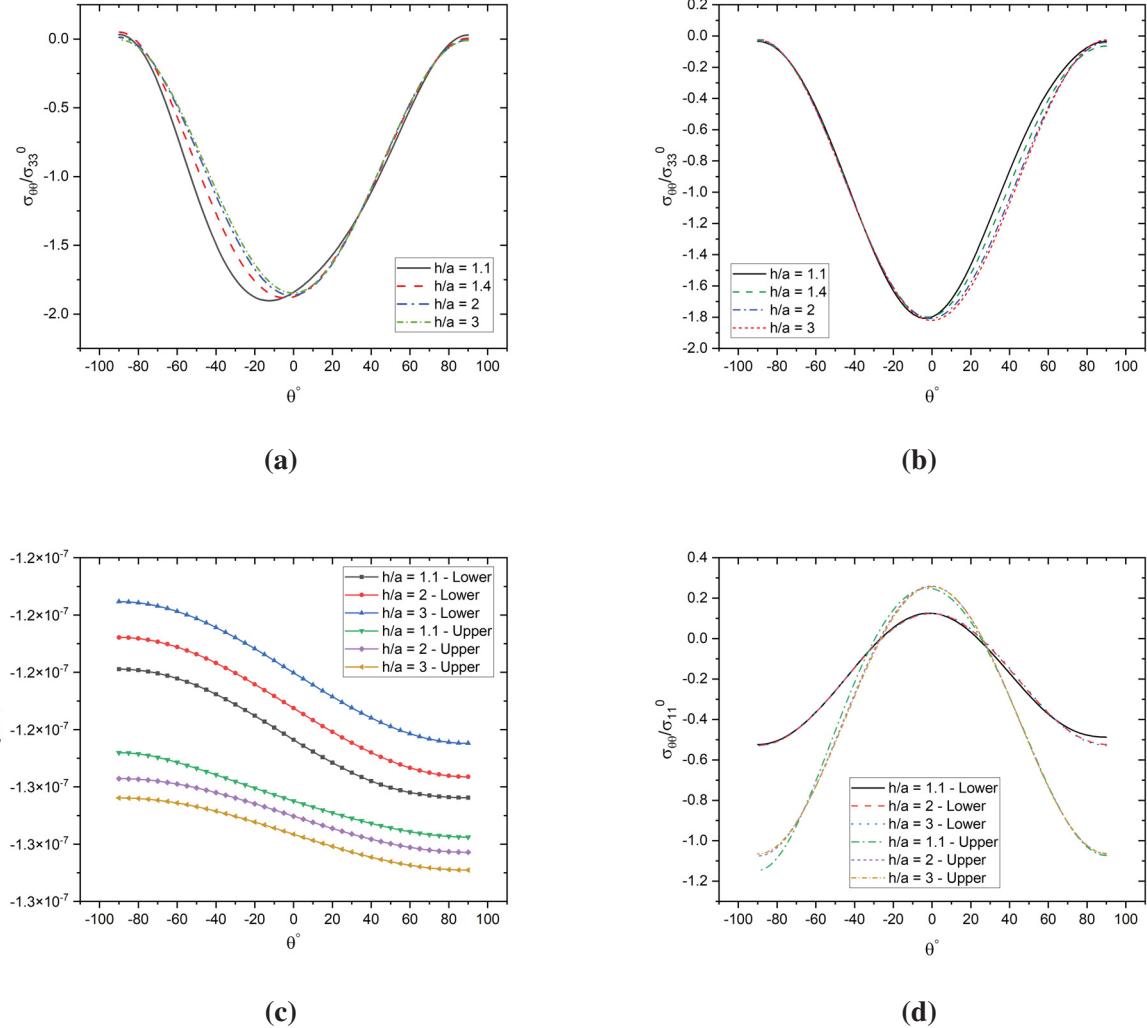
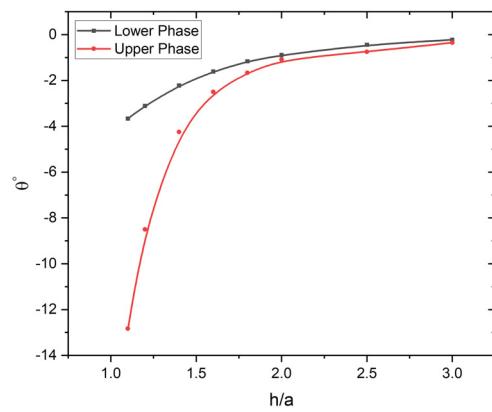
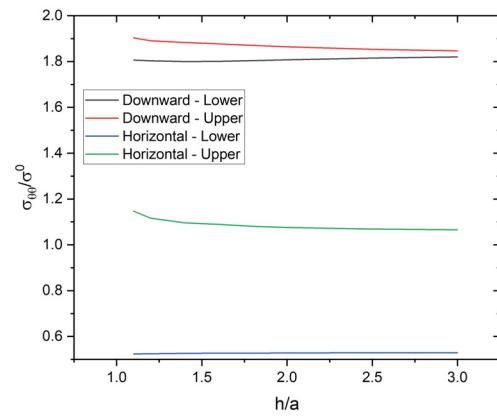


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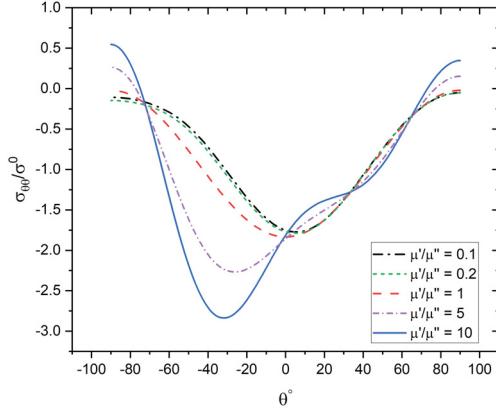


(a)

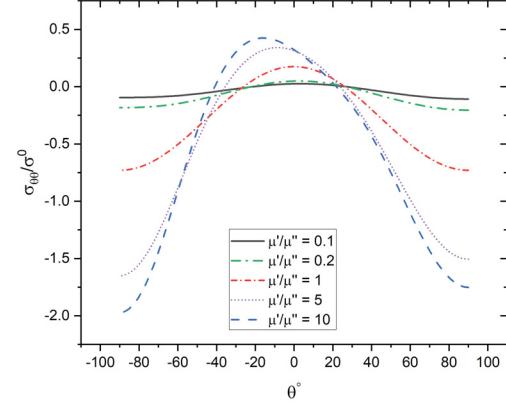


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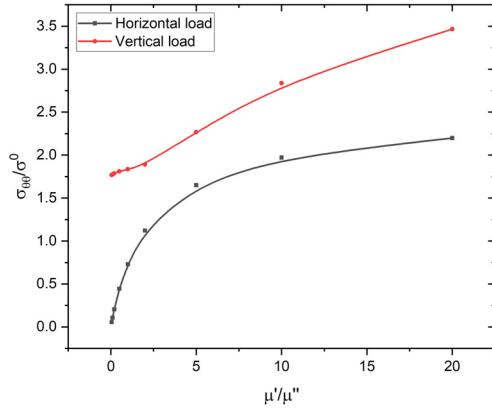
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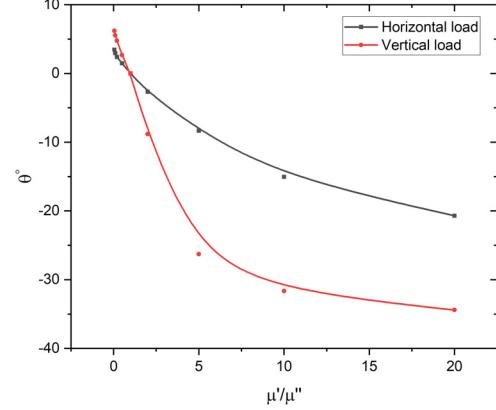
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(b)



(c)



(d)

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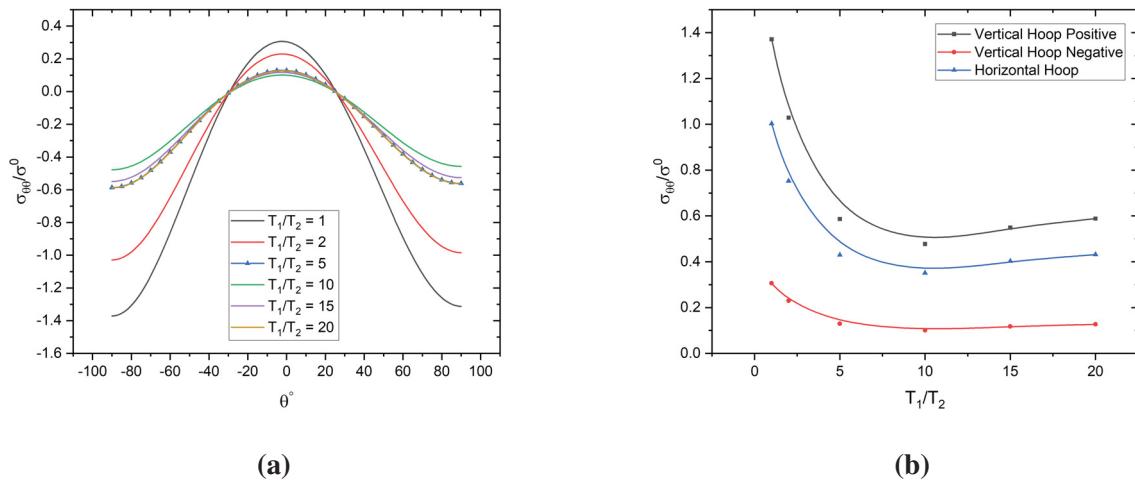


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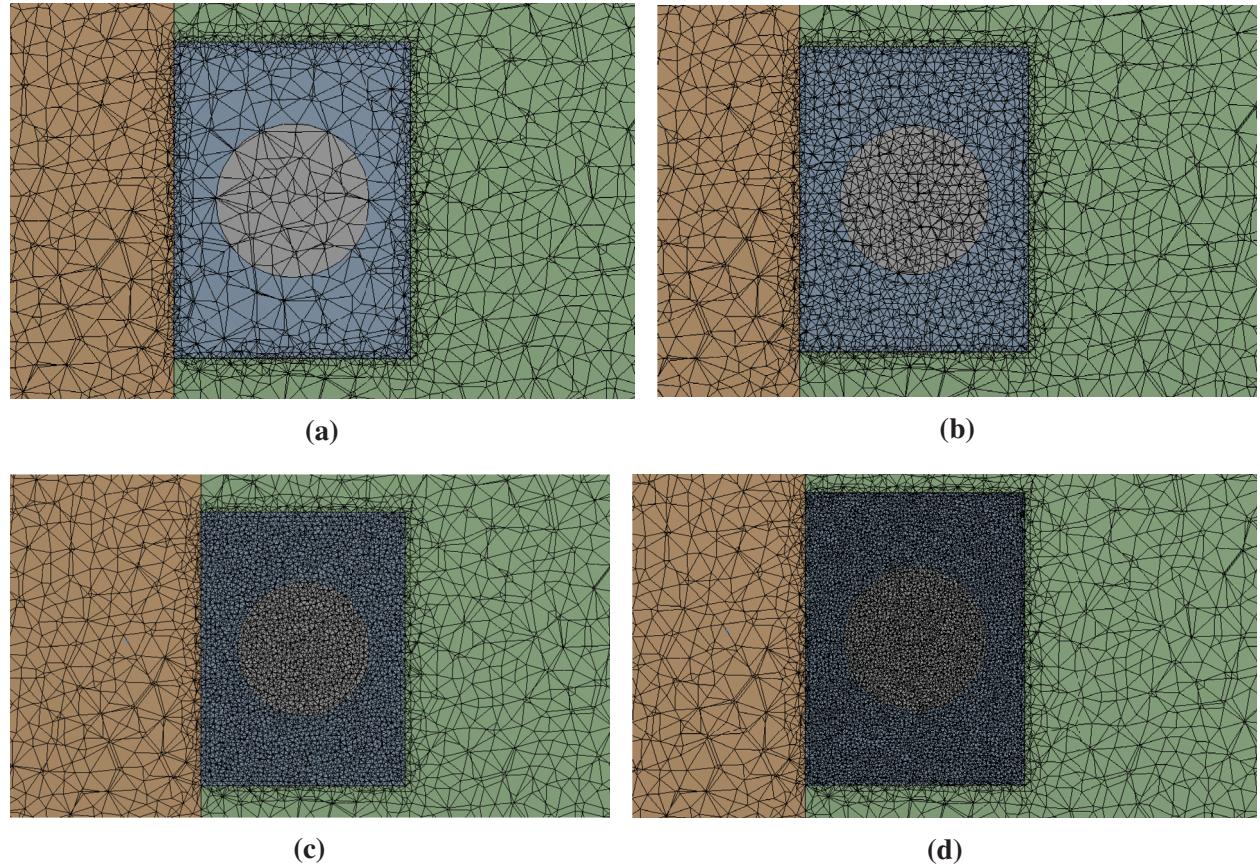
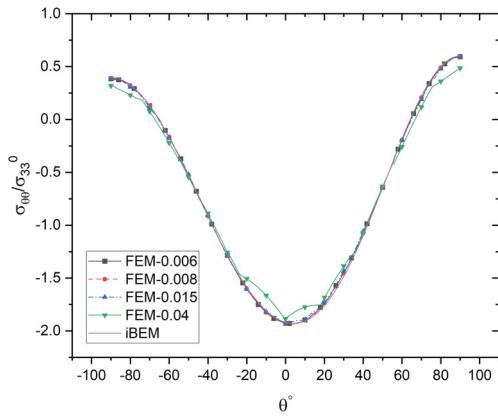
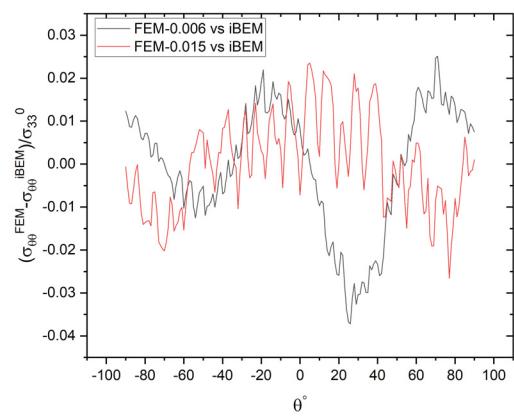


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(a)



(b)

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