

# Stress concentration of a micro-void embedded in a bi-layered material considering the boundary effects

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## ABSTRACT

This paper extends the Eshelby's problem of one inhomogeneity embedded in a homogeneous infinite domain to a bi-material infinite domain. The equivalent inclusion method (EIM) is used to simulate the inhomogeneity by an inclusion with a polynomial eigenstrain. The fundamental solution of a point force in a bi-material is used to formulate the domain integral over the inclusion. For a finite bi-material domain, the boundary integral equation (BIE) takes into account the boundary responses by a single domain instead of utilizing the conventional multi-region BIE scheme. The EIM can similarly be used, and the elastic field can be obtained with tailorable accuracy based on the order of the polynomial eigenstrain. The algorithm is particularly suitable to simulate a defect in thin film/substrate systems or other similar bi-layered materials. Particularly, the stress concentration of a microvoid embedded in a bi-layered solar panel is investigated. The size and location of the void referred to the interface exhibits considerable effects on the stress concentration factor. Numerical case studies demonstrate the effectiveness and accuracy of the algorithm, and parametric studies show the boundary effects on the stress concentration of a microvoid in a finite bi-material under a uniform far field strain.

## INTRODUCTION

The multi-layered systems have been widely utilized in versatile engineering and construction aspects, such as asphalt pavements (Yin and Prieto-Muñoz 2013), thin film surface coatings with protective and functional purposes (Abu-Thabit 2020; Ruys and Sutton 2021), and composite laminates (Anbusagar et al. 2015; Rana and Figueiro 2016). However, the defects during manufacturing process, such as air voids, may significantly jeopardize the reliability and lifetime of the overall bi-material system. Sengab and Talreja (Sengab and Talreja 2016) summarized two main sources of those defects, (i) impurities and air evaporating during curing process; (ii) entrapment of air during manufacturing process. For example, in solar panel manufacturing (Yin et al. 2022b), any voids in a solar panel may disturb the light transmission, form hot spots under strong sunlight, and cause microcracks and failure of the solar panel. Even for a homogeneous encapsulate layer, the defects can cause stress concentrations leading to cracks and failure. Especially, when defects are close to the interface  $S$ , the discontinuity of stress across the interface augments the stress concentration and singularity effects. Therefore, high fidelity stress analysis may provide more insights to understand this phenomena.

To investigate the stress transfer between layered materials, Stoney (Stoney 1909) proposed an approach with plate system assumptions, such as thick substrate to ignore bending stiffness of thin film, equal twist curvatures and spatially constant surviving stress (Ngo et al. 2007). Because the strong assumptions violate practical applications, several subsequent extensions have been proposed to relax them (Wikström et al. 1999; Park and Suresh 2000). However, the Stoney theory ignores the shear stress transfer, modified theories (Haftbaradaran et al. 2012; Zhang et al. 2021) have been proposed to consider the interfacial sliding effects. Since the above previous works assume two dissimilar homogeneous material phases, therefore those models cannot provide accurate analysis for bi-material system with defects. Regarding influence brought by micro defects, Katnam et al. (Katnam et al. 2011) investigates the formulation of air voids with two adhesive mixing techniques and used X-ray to detect and evaluate porosity; Omairey et al. (Omairey et al. 2021) summarized several failure modes of adhesive joints of composites, where adhesive defects and substrate defects

52 can cause high stress concentration leading to failures; Mishnaevsky (Mishnaevsky 2022) reported  
53 that even the defects in adhesive of the wind turbine blade may not dramatically disturb overall  
54 stress field, the local high stress concentration will lead to crack initiation.

55 To understand the effects of defects, the equivalent inclusion method (EIM) (Eshelby 1957;  
56 Eshelby 1959) was proposed to replace the defects with same matrix material along with inelastic  
57 strain, eigenstrain, to be determined by equivalent stress equations. With solved eigenstrain, the  
58 elastic fields can be acquired through superposition of initial fields and disturbance of eigenstrain,  
59 which is domain integral of fundamental solution over the inhomogeneity. Thanks to the versatility  
60 of fundamental solutions, EIM has been widely extended to other problems, such as heat conduc-  
61 tion (Hatta and Taya 1986; Wu et al. 2021), dynamic elasticity (Song and Yin 2018), etc. The  
62 Eshelby's solution of one inhomogeneity over the infinite matrix ignores the interactions among  
63 inhomogeneities themselves and the boundary (Liu and Yin 2014; Wu and Yin 2021). Based  
64 on EIM, pioneers developed micromechanical models, such as the dilute, Mori-Tanaka (Mori and  
65 Tanaka 1973; Kanit et al. 2003; Yin and Zhao 2016) and self-consistent models (Hershey 1954;  
66 Kroner 1958), which provides effective mechanical properties of composites and bridge the mi-  
67 crostructure and macroscopic behaviors. Other contributions on the homogenization schemes from  
68 linear elasticity to nonlinear rate-dependent problems can be found in (Zaoui 2002).

69 In the literature, several previous works explore the stress intensity factors (SIFs) caused by  
70 cracks and interfacial defects. Rather than using the conventional FEM, Treifi and Oyadiji (Treifi and  
71 Oyadiji 2013) proposed a fractal-like FEM to investigate SIFs of notch bodies with displacement  
72 interpolation functions. Bouhala et al. (Bouhala et al. 2013) developed crack-tip enrichment  
73 functions with extended FEM (XFEM) to study SIFs for cracks terminating at interface of bi-  
74 material; Pathak et al. (Pathak et al. 2011) combined element free Galerkin method and XFEM  
75 on crack interaction problems. Kaddouri et al. (Kaddouri et al. 2006) studied a practical case with  
76 couple metal-ceramic on factors associated with perpendicular cracks to bi-material interface, such  
77 as distance of crack-tip to the interface.

78 As for boundary element method (BEM) with Kelvin's solution, the multi-region scheme is

79 commonly used that interface of inhomogeneity and interface of bi-material require surface mesh  
80 and continuity equations are built to formulate the boundary value problem (BVP) (Beer et al. 2008;  
81 Liu et al. 2011). Fortunately, the continuity equations on interfaces of bi-material or multi-layered  
82 material can be mathematically considered with the fundamental solutions. Walpole (Walpole 1996)  
83 derived fundamental solutions to two-jointed dissimilar isotropic half-spaces through method of  
84 images; Yue (Yue 1995) proposed Yue's treatment, which is a generalized Kelvin's solution to  
85 multi-layered material; and other contributions in the literature can be found in review (Liu et al.  
86 2011). Xiao et al. (Xiao et al. 2019) applied Yue's treatment (Yue 2015) in BEM to investigate semi-  
87 infinite transversely isotropic domain, especially for simulation of rocks. For a bi-material system,  
88 Yue's treatment can be reduced to explicit formulae, and Wu et al. (Wu et al. 2022) completed  
89 Walpole's solution and applied it for analysis of bi-material system. The above works, though,  
90 save efforts in discretizing bi-material interface, when the number of inhomogeneity increases,  
91 dimension scale varies or close to boundary, the multi-region scheme on inhomogeneity requires  
92 both considerable computational resources and preparation process.

93 In our recent work (Yin et al. 2022a), the algorithm of inclusion-based boundary element method  
94 (iBEM) is designed to handle drastically increase of DOFs in simulation of composites. Using  
95 technique of fundamental solution, the material mismatch between inhomogeneity and matrix can  
96 be simulated with eigenstrain field without mesh of subdomains. Compared with Eshelby's uniform  
97 eigenstrain assumption in (Eshelby 1957), eigenstrain is presented by Taylor series expansion at  
98 the centroid of inhomogeneity (Mura 1987). The algorithm iBEM combines the BEM and EIM,  
99 where the boundary effects and interactions between inhomogeneities are considered in BEM global  
100 matrix and equivalent stress equations, respectively. Solving the system of linear equations, the  
101 boundary responses and eigenstrain field can be obtained. The advantages of iBEM are: (i) for  
102 each inhomogeneity, the number of DOFs is fixed, 6, 24, 60 for uniform, linear and quadratic order,  
103 respectively; (ii) avoid any subdomain mesh, including bi-material interface and inhomogeneity  
104 and potential numerical errors brought by them; (iii) the merit of BEM and fundamental solution  
105 is retained that internal fields are expressed in boundary and domain integrals.

106 This paper aims to perform elastic analysis of an inhomogeneity embedded a bi-material system  
 107 through the single-domain iBEM implemented with bi-material fundamental solution. In the  
 108 following, the problem of a bi-material system with an inhomogeneity is firstly proposed, and then  
 109 the fundamental solutions of bi-material, domain integrals over spherical inhomogeneity and global  
 110 matrix of iBEM are introduced. Subsequently, the aforementioned iBEM is verified with FEM for  
 111 a benchmark comparison. Applying the solution to a solar module containing a glass layer over  
 112 a concrete panel, when a microvoid is embedded in the substrate, the SIFs are investigated with  
 113 various distance to the bi-material interface. Finally, some conclusive remarks are discussed.

## 114 **PROBLEM STATEMENT**

115 Consider a domain  $\mathcal{D}$  embedded with one subdomain  $\Omega_1$  is composed of two dissimilar  
 116 isotropic domain, where the upper phase  $\mathcal{D}^+$  and the lower phase  $\mathcal{D}^-$  generally exhibits different  
 117 material properties  $C'$  and  $C''$ , respectively. For instance, the stiffness tensor of  $\mathcal{D}^+$  is  $C'_{ijmn} =$   
 118  $\lambda' \delta_{ij} \delta_{mn} + \mu' (\delta_{im} \delta_{jn} + \delta_{in} \delta_{jm})$ , where  $\lambda'$  and  $\mu'$  are lame constants of  $\mathcal{D}^+$ . The dimensions of the  
 119 bi-material system are defined in Fig.1 that, (i)  $T_1$  and  $T_2$  are thickness of  $\mathcal{D}^+$  and  $\mathcal{D}^-$ , respectively;  
 120 (ii)  $l$  and  $b$  are length and width. Shown in Fig. 1,  $\mathcal{D}$  is subjected to prescribed boundary conditions,  
 121 where  $t_i$  and  $u_i$  represents surface traction and displacement, respectively. Without the loss of any  
 122 generality, the bi-material interface  $S$  is chosen as parallel to plane  $x_1 - x_2$  at  $x_3 = 0$ . In the  
 123 following, two assumptions are made: (i) the embedded subdomain is filled with an isotropic  
 124 material and its stiffness tensor  $C^I_{ijmn}$ ; (ii) the bi-material and subdomain interfaces are perfect  
 125 without any debonding behavior, which satisfy the continuity equations on both displacement and  
 126 normal traction shown in Eq. (1).

$$127 \quad u_i(\mathbf{x}^+) = u_i(\mathbf{x}^-), \quad \sigma_{ij}(\mathbf{x}^+)n_j(\mathbf{x}^+) = \sigma_{ij}(\mathbf{x}^-)n_j(\mathbf{x}^-) \quad (1)$$

128 where "+" and "-" represents the inward and outward side of the bi-material interface  $S$  or subdomain  
 129 interface, respectively;  $\mathbf{n}$  is the unit surface normal vector. Subsequently, the BVP can be formulated  
 130 and it can be solved through the conventional multi-region scheme with interface mesh, which

131 commonly demands high computational costs due to the singularity and discontinuity along the  
 132 interface. This paper proposes an alternative method in which the fundamental solution for a bi-  
 133 material infinite domain is directly applied to Eshelby's equivalent inclusion method and boundary  
 134 integral method.

## 135 FORMULATION

### 136 Fundamental Solutions

137 Considering a two-jointed dissimilar half-spaces, the displacements at field point  $\mathbf{x}$  can be  
 138 expressed through the superposition of Kelvin's solution and image terms (Walpole 1996). The  
 139 Green's function define the displacement response of any field point  $\mathbf{x}$  caused by unit excitation at  
 140 source point  $\mathbf{x}'$ . Given a unit concentrated force  $f_j(\mathbf{x}') = n_j \delta(\mathbf{x}')$  ( $\delta(\mathbf{x}')$  is the Dirac delta function)  
 141 in the direction  $\mathbf{n}$ , the displacement variation can be expressed as,

$$142 \quad u_i(\mathbf{x}') = G_{ij}(\mathbf{x}, \mathbf{x}') f_j(\mathbf{x}') \quad (2)$$

143 Due to the position of image terms, the fundamental solution differs whether source point  $\mathbf{x}'$  and  
 144 field point  $\mathbf{x}$  are in the same material phase.

$$G_{ij}^y(\mathbf{x}, \mathbf{x}') = \frac{1}{4\pi\mu^w} \begin{cases} \left( \delta_{ij}\phi - \frac{\psi_{,ij}}{4(1-\nu^w)} \right) + A^y \bar{\phi} \delta_{ij} + \chi B^y (\delta_{i3}\delta_{jk} - \delta_{ik}\delta_{j3}) \bar{\alpha}_{,k}^y \\ - C^y x_3 [Q_J \bar{\psi}_{,ij3} + 4(1-\nu^w) \delta_{j3} \bar{\phi}_{,i} + 2(1-2\nu^w) \delta_{i3} Q_J \bar{\phi}_{,j} - Q_J x_3 \bar{\phi}_{,ij}] & x'_3 x_3 \geq 0 \\ - D^y Q_I Q_J \bar{\psi}_{,ji} - (G^y + B^y) Q_J \bar{\beta}_{,ij}^y \\ \left( \delta_{ij}\phi - \frac{\psi_{,ij}}{4(1-\nu^w)} \right) + A^y \phi \delta_{ij} + \chi B^y (\delta_{i3}\delta_{jk} - \delta_{ik}\delta_{j3}) \alpha_{,k}^y & x'_3 x_3 < 0 \\ - D^y \psi_{,ij} - \chi x_3 F^y \alpha_{,ij}^y - (G^y + B^y) Q_I \beta_{,ji}^y \end{cases} \quad (3)$$

145 where the coefficient  $\chi = 1$ , and superscripts  $w = u$  and  $y = u$  when  $x'_3 \geq 0$ ; and  $\chi = -1$ ,  $w = l$   
 146 and  $y = l$  when  $x'_3 < 0$ ;  $\mathbf{Q} = (1, 1, -1)$  is created for image terms through the interface  $S$  and the  
 148

149 dummy index rule does not apply to capitalized ones;  $\psi = |\mathbf{x} - \mathbf{x}'|$  is the Garlerkin distance vector  
 150 and  $\phi = \psi^{-1}; \frac{\delta_{ij}\phi}{4\pi\mu^w} - \frac{\psi_{,ij}}{16\pi(1-\nu^w)}$  is the Kelvin's solution of infinite space;  $\overline{(\cdot)}$  stands for image source  
 151 points, such that  $\overline{\psi} = |\mathbf{x} - \overline{\mathbf{x}}'|$  and  $\overline{\mathbf{x}}' = (x'_1, x'_2, -x'_3)$ ;  $A^u - G^u$  are material constants related to the  
 152 upper phase  $\mathcal{D}^+$ ,

$$\begin{aligned}
 A^u &= \frac{\mu' - \mu''}{\mu' + \mu''}, & B^u &= \frac{2\mu'(1 - 2\nu')(\mu' - \mu'')}{(\mu' + \mu'')(\mu' + \mu''(3 - 4\nu'))} \\
 C^u &= \frac{\mu' - \mu''}{2(1 - \nu')(\mu' + (3 - 4\nu')\mu'')}, & D^u &= \frac{3 - 4\nu'}{2}C \\
 F^u &= \frac{2\mu'(\mu'(1 - 2\nu'') - \mu''(1 - 2\nu'))}{(\mu' + \mu''(3 - 4\nu'))(\mu'' + \mu'(3 - 4\nu''))} \\
 G^u &= \frac{\mu'(\mu''(1 - 2\nu'')(3 - 4\nu') - \mu'(1 - 2\nu')(3 - 4\nu'))}{(\mu' + \mu''(3 - 4\nu'))(\mu'' + \mu'(3 - 4\nu''))}
 \end{aligned} \tag{4}$$

154 Similarly, coefficients  $A^l - G^l$  can be obtained by switching two material phases, i.e.  $A^l = \frac{\mu'' - \mu'}{\mu'' + \mu'}$ .

155 The other components in fundamental solution are listed below,

$$\begin{aligned}
 \alpha^u &= \ln[x'_3 - x_3 + \psi], & \overline{\alpha}^u &= \ln[x'_3 + x_3 + \overline{\psi}] \\
 \beta^u &= (x'_3 - x_3)\alpha^u - \psi, & \overline{\beta}^u &= (x'_3 + x_3)\overline{\alpha}^u - \overline{\psi} \\
 \alpha^l &= \ln[-x'_3 + x_3 + \psi], & \overline{\alpha}^l &= \ln[-x'_3 - x_3 + \overline{\psi}] \\
 \beta^l &= (-x'_3 + x_3)\alpha^l - \psi, & \overline{\beta}^l &= (-x'_3 - x_3)\overline{\alpha}^l - \overline{\psi}
 \end{aligned} \tag{5}$$

157 The  $\alpha$  functions are also known as Bousinesq's displacement potentials, which are elaborated in  
 158 Section "Domain Integral". According to Eqs.(2-4), when  $C' = C''$ , the coefficients  $A - G$  vanishes  
 159 and the fundamental solution reduce to Kelvin's solution; When one material phase exhibits zero  
 160 stiffness, the fundamental solution reduces to the Mindlin's problem (Yin et al. 2022a).

### 161 One Inclusion in Two-jointed Dissimilar Half-spaces

162 Consider an infinite domain  $\mathcal{D}$  composed of two-jointed dissimilar half-spaces, and one sub-  
 163 domain  $\Omega_I$  is subjected to eigenstrain  $\varepsilon_{ij}^*(\mathbf{x})$ . Notice that although in Eshelby's work (Eshelby  
 164 1957; Eshelby 1959) the eigenstrain is constant over the subdomain, the eigenstrain indeed can  
 165 vary spatially. Mura (Mura 1987) found that under interactions of inhomogeneities, the eigenstrain

166 is not uniform anymore, and thus proposed to use the Taylor series expansion at the center of the  
 167 subdomain to approximate eigenstrain as Eq.(6),

$$168 \quad \varepsilon_{ij}^*(\mathbf{x}) = \varepsilon_{ij}^{I0*} + (x_p - x_p^{Ic})\varepsilon_{ijp}^{I1*} + (x_p - x_p^{Ic})(x_q - x_q^{Ic})\varepsilon_{ijpq}^{I2*} \quad (6)$$

where  $\mathbf{x}^{Ic}$  is centroid of  $\Omega_I$  subdomain;  $\varepsilon^{I0*}$ ,  $\varepsilon^{I1*}$  and  $\varepsilon^{I2*}$  are uniform, linear and quadratic components of polynomial to approximate the eigenstrain. The disturbance displacement and strain field caused by eigenstrain can be obtained through the technique of Green's function as follows:

$$u_i(\mathbf{x}) = \int_{\Omega_I} \frac{\partial G_{ij}(\mathbf{x}, \mathbf{x}')}{\partial x'_m} \varepsilon_{kl}^*(\mathbf{x}') C_{jmkl}(\mathbf{x}') dV(\mathbf{x}') = g_{ikl}\varepsilon_{kl}^{I0*} + g_{iklp}\varepsilon_{klp}^{I1*} + g_{iklpq}\varepsilon_{klpq}^{I2*} \quad (7)$$

$$169 \quad \varepsilon_{ij}(\mathbf{x}) = S_{ijkl}\varepsilon_{kl}^{I0*} + S_{ijklp}\varepsilon_{klp}^{I1*} + S_{ijklpq}\varepsilon_{klpq}^{I2*} \quad (8)$$

171 where  $g_{iklpq\dots} = \int_{\Omega_I} G_{ij,m'} C_{jmkl}(x'_p - x_p^{Ic})(x'_q - x_q^{Ic}) dV(\mathbf{x}')$  is Eshelby's tensor for displacement;  
 172  $S_{pq..ijkl} = \frac{g_{iklpq\dots j} + g_{jklpq\dots i}}{2}$  is Eshelby's tensor for strain; indices  $p, q$  mean that polynomial-  
 173 form terms are involved, i.e  $\psi_p = (x'_p - x_p^{Ic})\psi$ . Notice that comparing with Kelvin's solution,  
 174  $G_{ij,m'} = -G_{ij,m}$  does not hold for bi-material fundamental solution and the partial derivatives are  
 175 provided in Appendix I.

## 176 Domain Integrals of Fundamental Solution with Polynomial Terms

177 In Eq.(3), the fundamental solution is obtained through superposition of Kelvin's solution  
 178 and image terms. As for Kelvin's solution, let  $\Phi$  and  $\Psi$  denote domain integrals of  $\phi$  and  $\psi$ ,  
 179 respectively. In 1891, Dyson (Dyson 1891) derived the general form domain integrals of elliptical  
 180 shell with various density functions. Later Moschovidis and Mura (Moschovidis and Mura 1975)  
 181 summarized Dyson's work and defined  $I$  and  $V$  functions to derive harmonic  $\Phi_{pq\dots}$  and biharmonic  
 182  $\Psi_{pq\dots}$  potentials, which will not be repeated below.

183 In terms of the image parts, let  $\Theta$  and  $\Lambda$  denote the domain integrals of  $\alpha$  and  $\beta$ , respectively.  
 184 Walpole (Walpole 1997) firstly present  $\Theta$  and  $\Lambda$  based on its definition of Bousinesq's displacement



185 potential and Liu et al. (Liu et al. 2015) extended them up to quadratic order ( $\Theta_{pq}, \Lambda_{pq}$ ). However,  
 186 the authors only considered cases of  $x'_3 > 0$ , thus in the following, we shall complete all cases of  
 187 domain integrals with more simplified definitions.

188 In Eq.(5), four types of  $\alpha$  and  $\beta$  functions are defined and the use of them is up to locations  
 189 of source and field points. Following definition of displacement potentials, the functions can be  
 190 rewritten as,

$$191 \begin{aligned} \alpha^u &= \int_{x_3}^{-\infty} \phi(x_1, x_2, t) dt & \bar{\alpha}^u &= \int_{\infty}^{x_3} \bar{\phi}(x_1, x_2, t) dt \\ \alpha^l &= \int_{\infty}^{x_3} \phi(x_1, x_2, t) dt & \bar{\alpha}^l &= \int_{x_3}^{-\infty} \bar{\phi}(x_1, x_2, t) dt \end{aligned} \quad (9)$$

192 where, only finite part of Eq.(9) are considered since infinite constant vanishes during partial  
 193 differentiation. Similarly,  $\beta$  functions can be written as Eq.(9) with the same integral limits but  
 194 switch integral functions from  $\phi$  or  $\bar{\phi}$  to  $\alpha$  or  $\bar{\alpha}$ , respectively. Because both  $\alpha$  and  $\beta$  functions are  
 195 defined through integrals along the third axis with respect to field point, one can interchange the  
 196 sequence of integral, taking  $\Theta^u$  as an example,

$$197 \begin{aligned} \Theta^u &= \int_{\Omega_I} \alpha^u dV(\mathbf{x}') = \int_{x_3}^{-\infty} \int_{\Omega_I} \phi dV(\mathbf{x}') dt = \int_{x_3}^{-\infty} \Phi(x_1, x_2, t) dt \\ \Lambda^u &= \int_{\Omega_I} \beta^u dV(\mathbf{x}') = \int_{x_3}^{-\infty} \int_{\Omega_I} \alpha^u dV(\mathbf{x}') dt = \int_{x_3}^{-\infty} \Theta^u(x_1, x_2, t) dt \end{aligned} \quad (10)$$

198 Notice that for the integral with respect to  $t$  along the third axis does not include any integral  
 199 points inside the subdomain, hence, only the exterior branch of  $\Phi$  and  $\Theta^u$  is retained. Following the  
 200 same fashion, other  $\Theta$ ,  $\Lambda$  and their polynomial involved functions can be derived. In the following,  
 201 their integrals are provided as below. The superscript  $s$  represents 4 types of functions defined in  
 202 Eq.(9), which is up to locations of source and field points.  $\phi^s = \bar{\phi}$ ,  $\psi^s = \bar{\psi}$  when  $x'_3 x_3 \geq 0$  and  
 203  $\phi^s = \phi$ ,  $\psi^s = \psi$  when  $x'_3 x_3 < 0$

204 *Uniform Domain Integrals  $\Theta$  and  $\Lambda$*

$$205 \Theta^s = \frac{4\pi a^3}{3} \alpha^s \quad \& \quad \Lambda^s = \frac{4\pi a^3}{3} \beta^s \quad (11)$$

206 *Linear Domain Integrals  $\Theta_p$  and  $\Lambda_p$*

$$207 \quad \Theta_p^s = \frac{4\pi a^5}{15} \begin{cases} -\alpha_{,p}^s & p \neq 3 \\ \phi^s & p = 3 \end{cases} \quad \& \quad \Lambda_p^s = \frac{4\pi a^5}{15} \begin{cases} -\beta_{,p}^s & p \neq 3 \\ \alpha^s & p = 3 \end{cases} \quad (12)$$

208 *Quadratic Domain Integrals  $\Theta_{pq}$  and  $\Lambda_{pq}$*

$$209 \quad \Theta_{pq}^s = \frac{4\pi a^5}{105} \begin{cases} -a^2 \alpha_{,q}^s (\alpha_{,p}^s + \ln[\psi^s]_{,p}) + \delta_{pq} [7\alpha^s + a^2 \phi^s \gamma^s] & p, q \neq 3 \\ -a^2 \phi_{,p}^s & p \neq 3, q = 3 \\ a^2 \phi_{,3}^s + 7\alpha^s & p = q = 3 \end{cases} \quad (13)$$

210 and

$$211 \quad \Lambda_{pq}^s = \frac{4\pi a^5}{105} \begin{cases} -a^2 (x_p - x_p^{Ic}) \gamma_{,q}^s + \delta_{pq} [7\beta^s - a^2 \gamma^s] & p, q \neq 3 \\ -a^2 \alpha_{,p}^s & p \neq 3, q = 3 \\ a^2 \alpha_{,3}^s + 7\beta^s & p = q = 3 \end{cases} \quad (14)$$

212 where  $\gamma^s$  is argument of the logarithmic function ( $\alpha^s$ ). When  $\alpha^s = \alpha^u$ , we can obtain  $\gamma^s = x'_3 - x_3 + \psi$ .

### 213 **One Inhomogeneity in a Bounded Bi-material Domain**

214 In the last subsection, the disturbance from polynomial-form eigenstrain can be obtained through  
 215 the explicit domain integral of fundamental solution over the spherical subdomain  $\Omega_I$ . Combining  
 216 with the conventional BEM, the elastic field is superposition of boundary responses with BIEs and  
 217 disturbance of eigenstrain with Eq.(7) and Eq.(8), and the displacement of arbitrary field point  $\mathbf{x}$   
 218 within  $\mathcal{D}^+$  can be expressed as,

$$219 \quad \begin{aligned} u_i(\mathbf{x}) &= - \int_{\partial \mathcal{D}^+} T_{ij}(\mathbf{x}, \mathbf{x}') u_j(\mathbf{x}') d\mathbf{x}' + \int_{\partial \mathcal{D}^+} G_{ij}(\mathbf{x}, \mathbf{x}') t_j(\mathbf{x}') d\mathbf{x}' + \int_{\Omega_I} \frac{\partial G_{ij}(\mathbf{x}, \mathbf{x}')}{\partial x'_m} \varepsilon_{kl}^*(\mathbf{x}') C_{jmkl}(\mathbf{x}') dV(\mathbf{x}') \\ &= - \sum_{e=1}^{NE} H_{ij} u_j^e + \sum_{e=1}^{NE} U_{ij} t_j^e + g_{ikl} \varepsilon_{kl}^{I0*} + g_{iklp} \varepsilon_{klp}^{I1*} + g_{iklpq} \varepsilon_{klpq}^{I2*} \end{aligned} \quad (15)$$

220 where  $G_{ij}$  and  $T_{ij} = \frac{C_{imkl}(\mathbf{x}')(G_{kj,l'} + G_{lj,k'})}{2} n_m(\mathbf{x}')$  are fundamental solution to displacement and  
 221 traction, respectively. With boundary surface mesh, the BIEs are expressed in a discretization form  
 222 (Beer et al. 2008) that  $H_{ij} = \int_{S_e} T_{mi}(\mathbf{x}, \mathbf{x}') N_{mj}(\mathbf{x}') dS$  and  $U_{ij} = \int_{S_e} U_{mi}(\mathbf{x}, \mathbf{x}') N_{mj}(\mathbf{x}') dS$ ;  $NE$  is  
 223 the number of elements; the superscript  $e$  represents nodal values of boundary displacements and  
 224 surface tractions in the  $e^{th}$  element. In Eq.(15), the interactions between subdomains and boundary  
 225 are involved to displacement of BEM. Since the continuity conditions of bi-material interface  $S$   
 226 has been analytically considered in the fundamental solutions, one can solve the boundary response  
 227 similar to a homogeneous matrix.

228 Unlike inclusion problems with prescribed eigenstrain, when the  $\Omega^I$  is filled with different  
 229 material  $C^I$ , the eigenstrain is yet to be determined with equivalent stress conditions. Mura (Mura  
 230 1987) proposed the conditions for polynomial-form eigenstrain to simulate material mismatch,

$$\begin{aligned}
 C_{ijkl}^w (\varepsilon_{kl}^b + \varepsilon'_{kl} - \varepsilon_{kl}^{I0*}) &= C_{ijkl}^I (\varepsilon_{kl}^b + \varepsilon'_{kl}) \\
 C_{ijkl}^w (\varepsilon_{kl,m}^b + \varepsilon'_{kl,m} - \varepsilon_{klm}^{I1*}) &= C_{ijkl}^I (\varepsilon_{kl,m}^b + \varepsilon'_{kl,m}) \\
 \frac{1}{2!} C_{ijkl}^w (\varepsilon_{kl,mn}^b + \varepsilon'_{kl,mn} - 2\varepsilon_{klmn}^{I2*}) &= \frac{1}{2!} C_{ijkl}^I (\varepsilon_{kl,mn}^b + \varepsilon'_{kl,mn})
 \end{aligned} \tag{16}$$

232 Because the inhomogeneity may be located in either the upper phase  $\mathcal{D}^+$  or the lower phase  $\mathcal{D}^-$ , in  
 233 Eq.(16), the superscript  $w ='$  when  $\Omega^I$  is located in  $\mathcal{D}^+$  and  $w =''$  when  $\Omega^I$  is located in  
 234  $\mathcal{D}^-$ .  $\varepsilon_{ij}^b$  is strain contributed by BIEs of boundary response with Eq.(20) in Appendix  
 235 I;  $\varepsilon'_{ij}$  is disturbed strain field expressed in Eq.(8). As indicated in Eq.(16), the interaction of  
 236 inhomogeneities are taken into account in equivalent stress conditions that one inhomogeneity can  
 237 disturb stress field of another inhomogeneity. Therefore, assembling the conventional BEM matrix,  
 238 collecting eigenstrain effects on boundary nodes and the stress equivalent equations, the iBEM  
 239 global system of linear equations will be shown in Eq. (19) of Appendix I.

240

## 241 Discussion and extension to ellipsoidal inhomogeneity

242 In this article, the above single-domain iBEM algorithm aims to save efforts in handling of trivial

243 procedures for domain discretization of the inhomogeneities. As for the industrial applications,  
244 Koenigsberger et al. (Königsberger et al. 2020) proposed a novel scheme on combination of  
245 Dvorak's transformation field analysis and Eshelby's method to investigate poro-elastic properties  
246 of cement paste, where the ellipsoidal Eshelby's tensor is introduced relating eigenstresses. In  
247 (Buchner et al. 2021) demonstrated experimental investigation on the effective elastic and thermal  
248 properties of clay bricks and various shapes of defects, i.e mesopores, quartz, are considered. Since  
249 the algorithm is based on the bi-material Green's function, and particularly the polynomial-form  
250 Eshelby's tensor used in equivalent stress conditions. Therefore, the limitations of the algorithm  
251 is the same limitation in the domain integral of the Green's function. Shown in Section "Domain  
252 integrals of fundamental solution with polynomial terms", the components of spherical Eshelby's  
253 tensor are derived by interchanging the integral sequence. Following the same fashion, (i) the  
254 domain integrals  $\Phi$  and  $\Psi$  over an ellipsoidal region can be found in (Dyson 1891) and (Moschovidis  
255 and Mura 1975); (ii) the domain integrals  $\Theta$  and  $\Lambda$  can be derived by partial integration along the  
256 third axis given integral limits. Notice that although Mura (Mura 1987) proposed the Taylor series  
257 expansion of eigenstrain to handle interactions of ellipsoidal inhomogeneities, when the ratios  
258 of axes of ellipsoid become too large / small, even quadratic eigenstrain terms may not provide  
259 accurate solution due to large variations of eigenstrain.

## 260 NUMERICAL VERIFICATION

261 The aforementioned algorithm is implemented to software package of iBEM to predict local  
262 fields of composites with prescribed boundary conditions. In order to validate the algorithm and  
263 illustrate how solutions with 3 orders of polynomial eigenstrain performs, a numerical case study  
264 with Robin's boundary condition is set up. Eshelby's solution of an inhomogeneity embedded in  
265 infinite space provides insight of uniform eigenstrain distribution. Actually, for a two-jointed half-  
266 spaces, if the inhomogeneity is far from the bi-material interface  $S$ , the image terms in fundamental  
267 solution Eq.(3) and its domain integral vanishes rapidly, which provides minor effects on stress  
268 disturbance. As a consequence, the elastic fields become similar to Eshelby's solution of infinite  
269 space. Therefore, the inhomogeneity will be placed close to the interface  $S$  to observe interfacial

270 effects on elastic fields. For spherical inhomogeneities, the intensity of interaction is commonly  
271 judged by the ratio  $d = h/a$ , where  $a$  is radius and  $h$  is distance from the centroid to interface.  
272 Following the definition of Fig.1, the dimensions and boundary conditions are set as: (i) the width  
273  $b$ , length  $l$  and two thickness  $T_1, T_2$  are set as 1 m; (ii) the inhomogeneity with radius  $a = 0.1$  m is  
274 placed in  $\mathcal{D}^-$ , where the distances  $h$  to interface  $S$  are  $h = 1.1a, 2a$  and  $3a$ ; (iii) shown in Fig.2,  
275 the top surface of  $\mathcal{D}^+$  is subjected uniform downward pressure 1 MPa; and the displacement of  
276 bottom surface of  $\mathcal{D}^-$  is constrained; all other four surfaces are free of traction. The shear modulus  
277 and Poisson's ratio for two material are  $\mu' = 0.4$  MPa,  $\nu' = 0.25$  and  $\mu'' = 0.8$  MPa and  $\nu'' = 0.1$ ,  
278 respectively.

279 Shown in Fig.3(a) and Fig.3(b), the elastic fields  $u_3$  and  $\sigma_{33}$  are compared with FEM with  
280 different ratio of distance. Because the disturbance by the inhomogeneity vanishes rapidly, the  
281  $[-3, 3]a$  around the inhomogeneity is considered. In simulation, 1000 quadrilateral boundary  
282 elements are used in iBEM and 2, 189, 461 tetrahedral elements are used in FEM. Regarding  $u_3$ ,  
283 the curves agree well with FEM except minor discrepancy is observed in the second branch of  
284 case  $h/a = 1.1$ . In terms of stress comparison, the main discrepancy between iBEM and FEM  
285 exists at the interface between inhomogeneity  $\Omega_I$  and matrix, which can be interpreted as intensive  
286 interfacial effects. Indicated in Fig.3(c), the stress concentration factors are compared in the vertical  
287 hoop of sphere and the maximum  $\sigma_{\theta\theta}$  changes with intensivity of interfacial effects, specifically, in  
288 the case  $h/a = 1.1$ , the angle shifts near 14 degree. As the  $h/a$  increases, both stress concentration  
289 and angle shift decreases, which will be elaborated in "Ratio of Distance". In Fig.3(d), the stress  
290  $\sigma_{33}$  ( $h/a = 1.2$ ) is compared along the center line among uniform, linear and quadratic series  
291 expansion. It is observed that the assumption of uniform distribution (Eshelby's solution) cannot  
292 predict the variation of elastic fields. The linear terms improves the accuracy but exhibit obvious  
293 discrepancy in the neighborhood of the inhomogeneity. Therefore, the introduction to quadratic  
294 term is necessary in improvement of accuracy of solution. **In the comparison among uniform, linear  
295 and quadratic order terms, as shown in Fig. 3(d), the accuracy of predictions improves with the  
296 increase of polynomial order. Such phenomenon indicates that due to the existence of interfacial**

297 effects, none of the numerical solution (uniform, linear and quadratic) is exact solution and so does  
298 the FEM. Back to Fig. 3(c), except the curve  $h/a = 1.1$ , the other two cases  $h/a = 2$  and  $h/a = 3$   
299 agree well with FEM. When the interfacial effects dominate, much more disturbance is observed,  
300 therefore, the disturbance also make the eigenstrain field more complex, which is the reason why  
301 even quadratic eigenstrains cannot provide adequately accuracy as iBEM exhibits an angle lag.

302 When the distance ratio  $h/a$  increases and interfacial effects decrease accordingly, the uniform  
303 term gradually dominates the solution. In such case, only uniform term alone can provide accurate  
304 results (Wu and Yin 2021). Although the iBEM algorithm has enabled three tailorable accuracy  
305 options, in the following, quadratic term is applied to ensure reliable and accurate analysis.

## 306 CASE STUDY OF STRESS CONCENTRATION FACTORS

307 In the previous section, the iBEM algorithm has been validated with FEM with three cases of  
308 different distance ratios in a bi-material system. In this section, the algorithm is further utilized to  
309 investigate elastic fields and stress concentration factors of a specific industrial application with a  
310 microvoid  $a = 2.5 \times 10^{-6}$ m in the sunslate. Because of the thin adhesive layer, the sunslate can be  
311 considered as a bi-material system with  $l = b = 0.1$  m of (i) glass ( $E' = 72$ GPa,  $\nu' = 0.2$ ,  $T_1$ ) and  
312 (ii) concrete panels ( $E'' = 36$ GPa,  $\nu'' = 0.2$ ,  $T_2$ ) (Yin et al. 2022b), as shown in Fig.1. The sunslate  
313 is subjected to downward pressure and thermal loads, which can be decomposed of superposition  
314 of free expansion and applied pressure for mismatch of thermal expansion coefficient (Yin and  
315 Prieto-Muñoz 2013), as shown in Fig.2. Since the free expansion does not result in variation of  
316 stress, this paper focus the second parts and the applied loads are set as 1MPa, which can be easily  
317 extended for other values of loads in linear elastic stage.

### 318 Ratio of Distance

319 The interfacial effects are judged through ratio of distance between the inhomogeneity and inter-  
320 face  $S$ . In this subsection, 8 ratio of distances are considered as  $h/a = 1.1, 1.2, 1.4, 1.6, 1.8, 2, 2.5, 3$   
321 and the microvoid is placed in lower and upper phase for overall 16 cases. In Fig.4(a) and Fig.4(b),  
322 the stresses  $\sigma_{\theta\theta}$  are plotted versus the vertical hoop angle  $\theta \in [-90^\circ, 90^\circ]$  when the microvoid is  
323 located in upper and lower phase, respectively. Comparing the angle of maximum stress, when

324  $h/a$  increases and interfacial effects decrease, the angle gradually decreases accordingly. If  $h/a$  is  
 325 large enough, the contribution from image terms in fundamental solutions vanishes and angle of  
 326 maximum stress changes as zero, which is similar to a infinite space problem but with disturbed  
 327 elastic fields due to the existence of the other phase. Obviously, the material mismatch influence  
 328 the stress concentration as well. Indicated in Fig.4(a) and Fig.4(b), when microvoid is embedded  
 329 in  $\mathcal{D}^+$ , the change of distance ratio exhibits larger variation of both angle shifts and stress concen-  
 330 tration factor. In Fig.5(a), the angle shift in  $\mathcal{D}^+$  is over 3 times than that of  $\mathcal{D}^-$ , which explains  
 331 larger material mismatch amplify interfacial effects. Fig.4(c) shows the displacement  $u_3$  of 6 cases  
 332 of microvoid under uniform downward pressure 1 MPa on the vertical hoop. When the distance  
 333 between microvoid and bottom surface decreases, the displacement  $u_3$  decreases accordingly due  
 334 to constraint of bottom surface. When angle  $\theta = -90^\circ$  and microvoid is embedded in  $\mathcal{D}^+$ , the  
 335 differences between case "h/a = 1.1 - Upper" and "h/a = 1.2 - Upper" is larger than other angles.  
 336 Such phenomenon is caused by interfacial effects associated with softer material concrete in  $\mathcal{D}^-$ .  
 337 Similarly, when angle  $\theta = 90^\circ$  and microvoid is embedded in  $\mathcal{D}^-$ , opposite trend is observed.  
 338 Shown in Fig.4(d), under horizontal pressure, the distance ratio has minor effects on stress concen-  
 339 tration and angle shifts, which can be interpreted as the direction of load is perpendicular to  $x_3$ .  
 340 Fig.5(b) exhibits the variation of stress concentration factor versus  $h/a$  with two types of loads and  
 341 positions of microvoid. When  $h/a$  increases, the stress concentration factors tends to be constant,  
 342 which is similar to Kelvin's problem.

### 343 **Ratio of Shear Moduli**

344 This subsection aims to present how ratio of shear moduli influence the stress concentration  
 345 caused by the microvoid. Discussed in "Ratio of distance" section, besides the interfacial effects,  
 346 the material mismatch also contribute to stress concentration. To avoid such factors, the microvoid  
 347 is placed in the upper phase  $\mathcal{D}^+$  with distance ratio  $h/a = 1.2$  and the material properties of the  
 348 upper phase remain the same ( $E' = 72\text{GPa}, \nu' = 0.2$ ). 9 ratios of shear moduli  $\frac{\mu'}{\mu''}$  are considered as  
 349 0.05, 0.1, 0.2, 0.5, 1, 2, 5, 10 and 20. Shown in Fig.6(a) and Fig.6(b), the stress concentration factors  
 350 increases with  $\mu'/\mu''$  and angle of maximum hoop stress shifts to negative range. When  $\mu'/\mu''$

351 is small, the stiffer lower phase restrict the displacements and vice versa; consequently, as partial  
352 derivatives of displacements, the stress concentration factor is smaller and angle of maximum stress  
353 shifts to positive range. In Fig.6(c), when  $\mu'/\mu'' = 1$ , the fundamental solution reduces to Kelvin's  
354 solution, the stress concentration factor under vertical load is close to 2. Fig.6(d) plots the angle of  
355 maximum stress versus ratios of shear moduli and vertical load case has larger variations because  
356 the loading direction is perpendicular to the interface  $S$ . The angles of maximum stress in two  
357 curves are close to  $5^\circ$ , which is caused by non-uniformity of strain field around the microvoid (not  
358 a far-field uniform strain).

### 359 **Ratio of Thickness**

360 This subsection aims to investigate thickness ratio effects on the stress concentration behavior  
361 caused by the microvoid. Following the "Ratio of shear moduli", the microvoid is placed in  $\mathcal{D}^+$  with  
362 distance ratio  $h/a = 1.2$  and 6 ratios 1, 2, 5, 10, 15 and 20 are selected. Fig.7(a) plots the vertical  
363 hoop stress with hoop angle. It is noticed that when  $T_1/T_2 \leq 10$  and the ratio increases, the negative  
364 stress concentration factor decreases and positive stress concentration factor increases obviously.  
365 However, from case 15 and case 20, the trend reverses that case 20 has similar stress distribution as  
366 case 5; case 15 exists between case 10 and 5. Such phenomenon is caused by the elastic behavior of  
367 thin panels or plates. When the thickness ratio  $T_1/T_2 \leq 10$ , under horizontal pressure, the primary  
368 forces for  $\mathcal{D}^+$  are bending and shear; as thickness ratio exceeds 10, the lateral shearing deformation  
369 become neglecting, which explains the maximum negative stress concentration increases due to  
370 elastic behavior change. Fig.7(b) indicates the trend of stress concentration factor discussed above.  
371 Under horizontal pressure, the horizontal hoop stress has similar trends as the other two curves.  
372 Notice that, unlike distance ratio and shear moduli ratio, the thickness ratio seldom change the  
373 angle of maximum negative/positive hoop stress. Because the angle shifts of maximum hoop stress  
374 is generally considered as interfacial effects, the smaller distance ratio, larger material mismatch  
375 and shear moduli ratio can augment such effects. However, the change of stress distribution with  
376 thickness ratios is mainly caused by the elastic behavior of the bi-material system itself.



## 377 **CONCLUSIONS**

378 The algorithm of the single domain inclusion-based boundary element method has been applied  
379 to investigate the elastic fields of bi-material system embedded with one microvoid. The algorithm  
380 has been verified by FEM with case study of void embedded in a two-cuboid bi-material system  
381 with different distance ratios. Along with the numerical verification, the iBEM with uniform, linear  
382 and quadratic terms are compared with FEM and provide tailorable accuracy upon readers' needs.  
383 Thanks to the fundamental solution of bi-material, and explicit domain integrals, the conventional  
384 boundary value problem with interface and subdomain can be solved similarly to a homogeneous  
385 solid. In this paper, the algorithm has been applied to study the stress concentration issues arisen  
386 in a Sunslate and parametric studies are conducted on effects brought by distance ratio, shear  
387 modulus ratio and thickness ratio and loading conditions. The algorithm is particularly suitable for  
388 understanding, designing and conducting virtual experiments on a thin-film system with potential  
389 defects.

## 390 **Data Availability Statement**

391 All data, models, or code that support the findings of this study are available from the corre-  
392 sponding author upon reasonable request.

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## APPENDIX I.

### Method of images in Walpole's solution and its modified form

This appendix subsection aims to provide details and discussion of fundamental solution by Walpole (Walpole 1996). For a perfect bounded bi-material interface, the continuity conditions in Eq. (1) on displacements and tractions need to be satisfied. The method of images is a typical mathematical tool to solve partial differential equations that a mirror image source is artificially created to handle the continuity conditions of the interface without extending the domain of function. Haberman (Haberman 2021) illustrated the method through semi-infinite solution to Poisson's equation in Chapter 9.5.8 and transient heat transfer in Chapter 11.5.3. Walpole (Walpole 1996) wrote the fundamental solution (displacement) in terms of two branches, which are determined by positions of field and source points. Subsequently, similar to Boussinesq's solution and Mindlin's problem, the potential functions are assumed as partial integration with respect to the third axis, where part of Kelvin's solution and nuclei of strain are applied. According to the continuity conditions, the coefficients associated with material constants can be determined.

The original form of bi-material elastic fundamental solution can be applied to investigated induced elastic fields caused by loads at any arbitrary interior point. However, its application to Eshelby's problem is not complicated and trivial as Eshelby's tensor leads domain integrals with the free source terms, i.e  $x'_3$ . It is possible to represent the free source terms with Garlekin's distance vector, for example, the domain integral of  $x'_3\phi$  can altered as  $\int_{\Omega}(x'_3 - x_3) + x_3\phi d\mathbf{x}' = -\Psi_{,3} + x_3\Phi$ , which simplifies the domain integral expressions.

### Partial Derivatives of Domain Integrals

It is noted that with image terms, partial differentiation process changes accordingly, say  $\bar{\alpha}''_{,i'} = -Q_I\bar{\alpha}''$ , we provide the first order derivative of fundamental solution to obtain Eshelby's tensor for displacement with quadratic order polynomial,

421 (1) When  $x'_3 x_3 \geq 0$ ,

$$\begin{aligned}
& 4\pi\mu^w \int_{\Omega} (x'_p - x_p^c)(x'_q - x_q^c) G_{ij,m'} dV(\mathbf{x}') \\
& = (-\delta_{ij}\Phi_{pq,m} + \frac{\Psi_{pq,ijm}}{4(1-\nu^w)}) - Q_P Q_Q \left\{ A^y \delta_{ij} Q_M \bar{\Phi}_{pq,m} - \chi B^y Q_M (\delta_{i3} \bar{\Theta}_{pq,jm}^y - \delta_{j3} \bar{\Theta}_{pq,im}^y) \right. \\
& - C^y x_3 Q_M \left[ -Q_J \bar{\Psi}_{pq,ij3m} - 4(1-\nu^w) \delta_{j3} \bar{\Phi}_{pq,im} - 2(1-2\nu^w) \delta_{i3} Q_J \bar{\Phi}_{pq,jm} + x_3 Q_J \bar{\Phi}_{pq,ijm} \right] \\
& \left. + Q_J Q_M \left[ D^c Q_I \bar{\Psi}_{pq,ijm} + (G^y + B^y) \bar{\Lambda}_{pq,ijm}^y \right] \right\}
\end{aligned} \tag{17}$$

423 (2) When  $x'_3 x_3 < 0$ ,

$$\begin{aligned}
& 4\pi\mu^w \int_{\Omega} (x'_p - x_p^c)(x'_q - x_q^c) G_{ij,m'} dV(\mathbf{x}') \\
& = (-\delta_{ij}\Phi_{pq,m} + \frac{\Psi_{pq,ijm}}{4(1-\nu^w)}) - A^y \delta_{ij} \Phi_{pq,m} - \chi B^y (\delta_{i3} \Theta_{pq,jm}^y - \delta_{j3} \Theta_{pq,im}^y) \\
& + D^y \Psi_{pq,ijm} + \chi x_3 F^y \Theta_{pq,ijm}^c + (G^y + B^y) Q_I \Lambda_{pq,ijm}^y
\end{aligned} \tag{18}$$

425 For other higher order derivatives to obtain Eshelby's tensor for strain is straightforward because  
426 the partial differentiation is with respect to  $x_n$  not  $x'_n$ .

### 427 Global Matrix of iBEM

$$\begin{aligned}
& \begin{bmatrix} \mathcal{H} + H & \dots & -g^{0I} & -g^{1I} & -g^{2I} & \dots \\ \vdots & \vdots & \vdots & \vdots & \vdots & \vdots \\ \Delta C^I H^{1I} & \dots & \Delta C^I S^{0I} & \Delta C^I S^{1I} & \Delta C^I S^{2I} & \dots \\ \Delta C^I H^{2I} & \dots & \Delta C^I S^{0I'} & \Delta C^I S^{1I'} & \Delta C^I S^{2I'} & \dots \\ \Delta C^I H^{3I} & \dots & \Delta C^I S^{0I''} & \Delta C^I S^{1I''} & \Delta C^I S^{2I''} & \dots \\ \vdots & \vdots & \vdots & \vdots & \vdots & \vdots \end{bmatrix}_{(3*NN+s*NI) \times (3*NN+s*NI)} \begin{bmatrix} u \\ \vdots \\ \varepsilon^{*0I} \\ \varepsilon^{*1I} \\ \varepsilon^{*2I} \\ \vdots \end{bmatrix}_{(3*NN+s*NI)} = \begin{bmatrix} U \\ \vdots \\ -\Delta C^I U^{1I} \\ -\Delta C^I U^{2I} \\ -\Delta C^I U^{3I} \\ \vdots \end{bmatrix}_{(3*NN+s*NI) \times (3*NN)} \begin{bmatrix} t \end{bmatrix}_{(3*NN)}
\end{aligned} \tag{19}$$

428 where  $s$  is determined order of polynomial by users, as 6, 24, 60 for uniform, linear and quadratic;  
 429  $\mathcal{H}$  is a diagonal matrix applied in conventional BEM to eliminate strong singularities with method  
 430 of rigid body motion for static problems;  $H$  and  $U$  are coefficients calculated by discretized BIE in  
 431 Eq.(15);  $g$  and  $S$  are Eshelby's tensor for displacement and strain (with uniform, linear and quadratic  
 432 order), respectively;  $H^{1I}, H^{2I}, H^{3I}$  and  $U^{1I}, U^{2I}, U^{3I}$  are coefficients calculated by discretized BIE  
 433 in Eq.(20) below for strain, first order partial derivative of strain and second order partial derivatives  
 434 of strain, respectively. They are used in equivalent stress conditions as Eq.(16).

$$\begin{aligned}
 U_{ijm}^{0I} &= \frac{1}{2} \int_{S_e} (H_{ik,j}(\mathbf{x}, \mathbf{x}') + H_{jk,i}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS \\
 H_{ijm}^{0I} &= \frac{1}{2} \int_{S_e} (T_{ik,j}(\mathbf{x}, \mathbf{x}') + T_{jk,i}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS
 \end{aligned}
 \tag{20}$$

436 Similarly, other higher order partial derivatives can be obtained, i.e,  $U_{ijmnr}^{2I} = \frac{1}{2} \int_{S_e} (H_{ik,jnr}(\mathbf{x}, \mathbf{x}') +$   
 437  $H_{jk,inr}(\mathbf{x}, \mathbf{x}')) N_{mk}(\mathbf{x}') dS.$

438

## APPENDIX II. COMPUTATIONAL TESTS BETWEEN IBEM AND FEM

This appendix section aims to provide details in computational efficiency and accuracy comparison between the proposed iBEM programmed by authors and FEM using a commercial software ANSYS. Regarding the accuracy and convergence, two aspects are discussed, (a) a convergence analysis of FEM through refinement of elements around the inhomogeneity; and (b) an error analysis concerning the differences between iBEM and FEM (convergent results). As for the efficiency, three main aspects are considered, (i) preparation stage, such as generation of geometric specification ,surface (iBEM) / volume (FEM) domain discretization; (ii) construction and solving process, such as computation of global matrix and solution time; and (iii) a comparison occupancy test of CPU and RAM in two methods.

Without the loss of any generality, this section follows "Numerical Verification" that all material properties are retained and the ratio of distance  $h/a = 1.5$ , where the inhomogeneity with radius  $a = 0.1$  m is located in the lower phase  $\mathcal{D}^-$ . For BEM surface mesh, 4-node bi-linear quadrilateral elements with 4 Gauss integral points are used with adaptive subdivision integration scheme following (Eberwien et al. 2005). Regarding FEM volume mesh, 10-node quadratic tetrahedral elements are used due to expected larger variation of displacements.

### Accuracy test

Shown as Fig. 8, in order to use fewer elements, the neighbor box (dimension  $0.4 \times 0.4 \times 0.3$  m) with the spherical inhomogeneity and the rest region of the matrix are treated with two mesh sizes. Four internal size steps are selected as 0.04, 0.015, 0.008 and 0.006 m, while the uniform external mesh size is 0.04 m. When the internal mesh size decreases, the number of elements increase accordingly as 41, 4906, 50, 9469, 114, 7928 and 232, 6729. Indicated in Fig. 9(a), the stress concentration factor  $\sigma_{\theta\theta}/\sigma_{33}^0$  gradually converges as the differences between size steps decrease. Particularly, size step 0.006 and 0.008 exhibit very minor discrepancies, and the two curves also agree well with results in iBEM. Considering the larger differences in Fig. 3(c), when the distance ratio  $h/a$  increase, the variation in eigenstrain decreases accordingly, which can provide good

466 predictions. Fig. 9(b) indicates the errors between iBEM and two FEM curves, although there are  
467 some differences, considering the maximum stress concentration factor is 1.94, the errors between  
468 two methods are acceptable.

469

### 470 **Efficiency test**

471 Shown as Table. 1, FEM and iBEM-quadratic used 2 and 8 cores, respectively. We keep  
472 the default setting of ANSYS, which limits participation of more cores, so that the solving time  
473 will be shorter. Since the solving process contains calculation, assignment of coefficients and  
474 matrix decomposition, the solution time is not linearly proportional to number of cores, because  
475 assignment of coefficients is partially a single-thread process. Although Table. 1 indicates iBEM  
476 package occupies all cores of the CPU, it uses "Eigen" library, where the process decomposition  
477 of matrix is not fully multi-core. The iBEM-quadratic exhibit apparent advantages on RAM usage  
478 over FEM, because the degree of freedom is a constant as 3,060. Regarding the mesh process,  
479 iBEM only require the surface mesh, which avoids trivial process on the inhomogeneity and its  
480 neighbor region as shown in Fig. 8 (d). In addition, FEM require specification of models, such  
481 as importing from AutoCAD or creating in its own geometry editors, which requires more efforts.  
482 Considering all above factors, iBEM would be an efficient and computational resource friendly  
483 scheme.

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**List of Tables**

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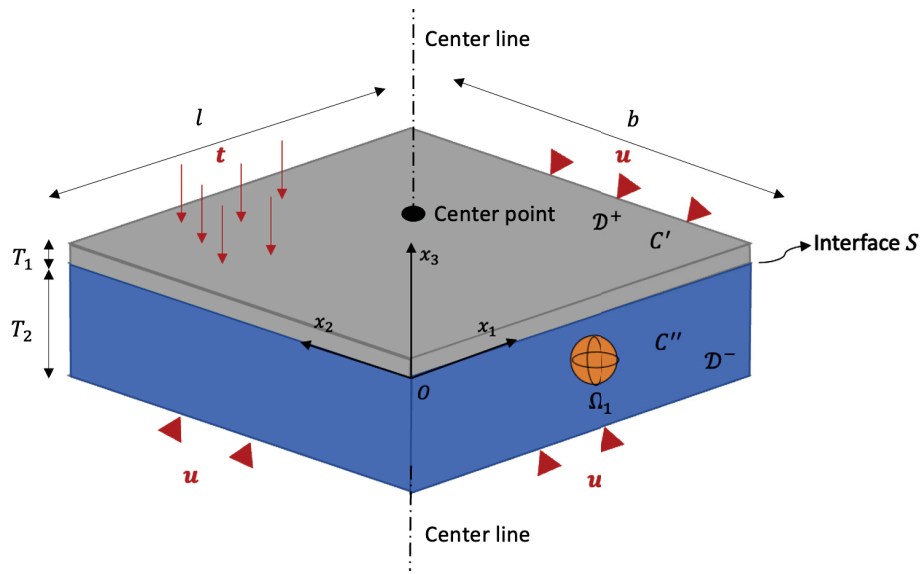
internal size steps . . . . . 29

**TABLE 1.** Comparison of efficiency among iBEM with quadratic terms and FEM with four internal size steps

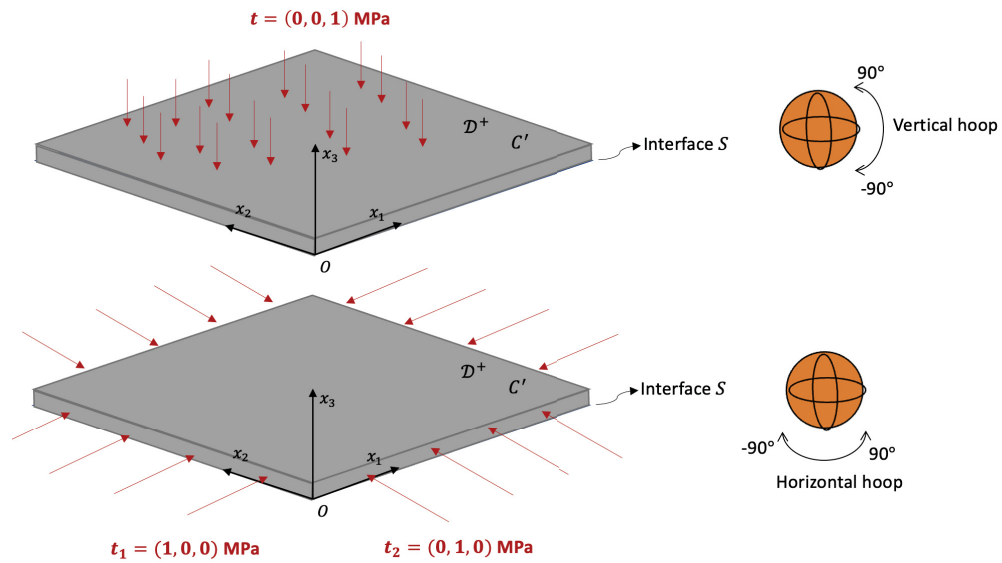
	CPU Usage (cores)	RAM Usage (Gb)	Mesh time (s)	Solving time (s)
FEM-0.04	2	2.22	17	48
FEM-0.015	2	3.51	27	58
FEM-0.008	2	7.65	39	133
FEM-0.006	2	12.60	72	326
iBEM-quadratic	8	0.58	5	10.2
CPU: i7-9700K (8 cores)		RAM: 2933mHz		

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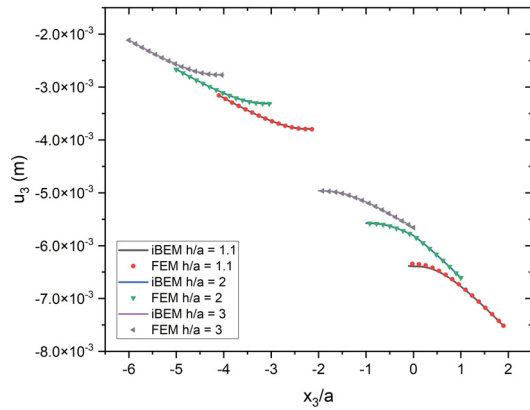


**Fig. 1.** Schematic plot of a bi-material system  $\mathcal{D}$  (a) subjected to mixed prescribed boundary conditions embedded with one inhomogeneity  $\Omega_I$

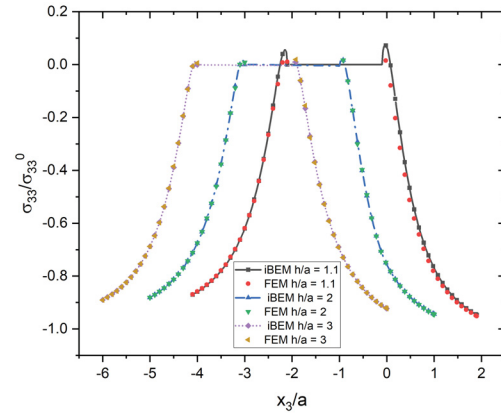


**Fig. 2.** Downward pressure and horizontal pressure loading cases and vertical/horizontal for stress comparison

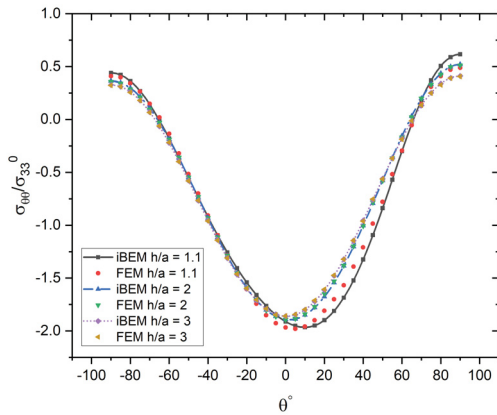




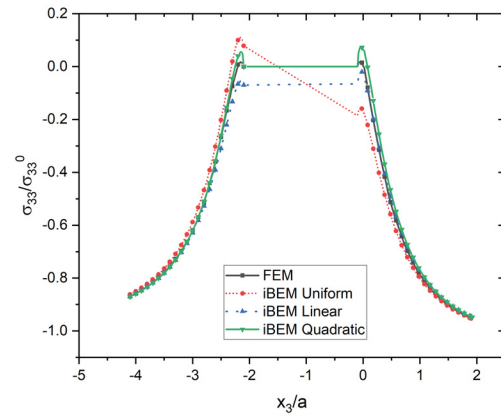
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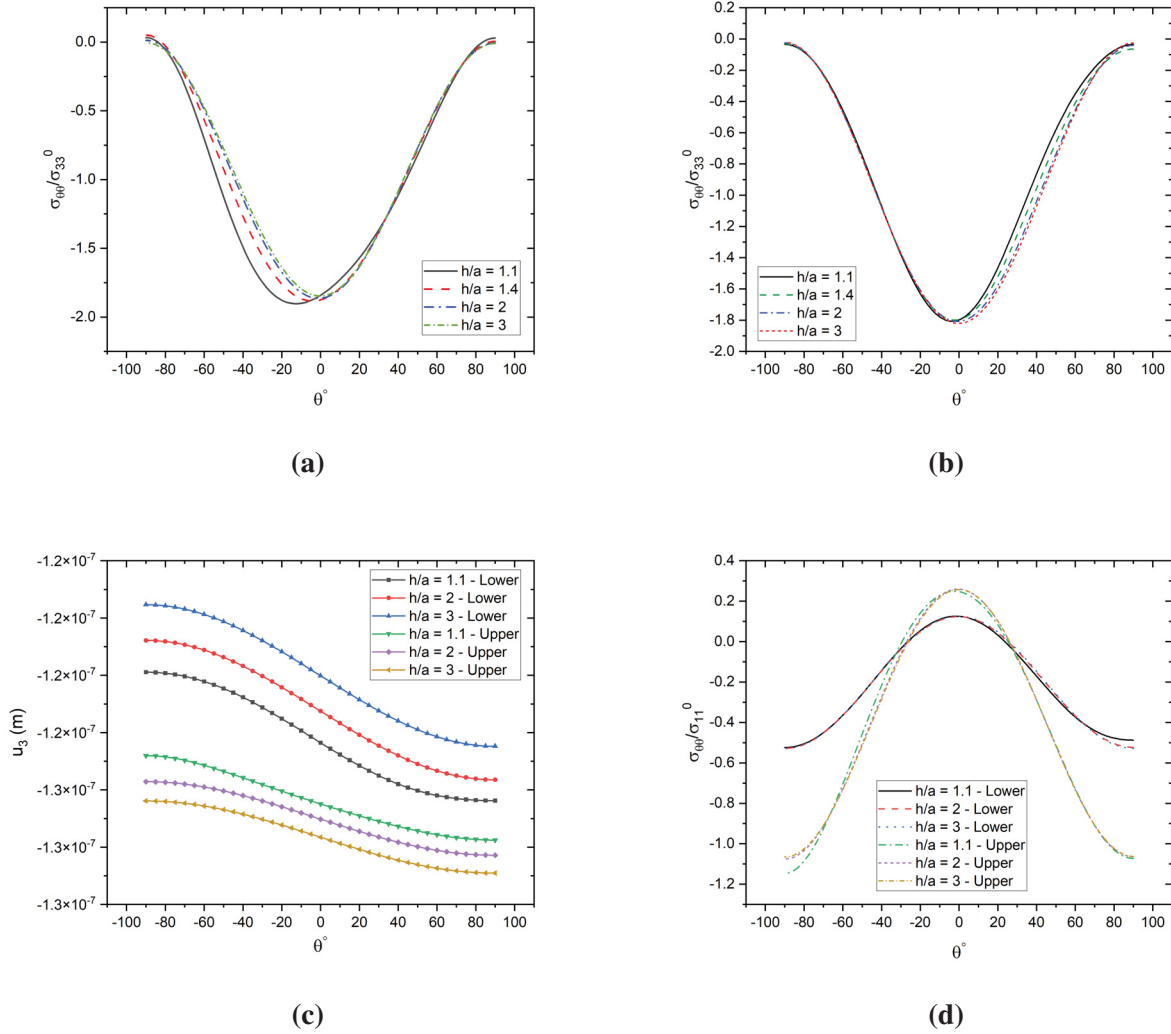


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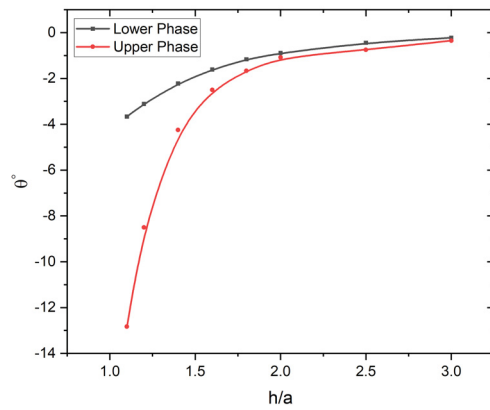


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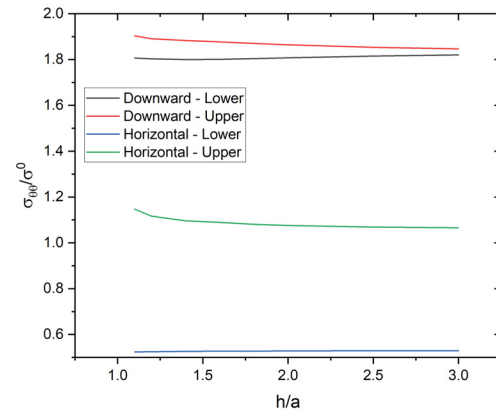
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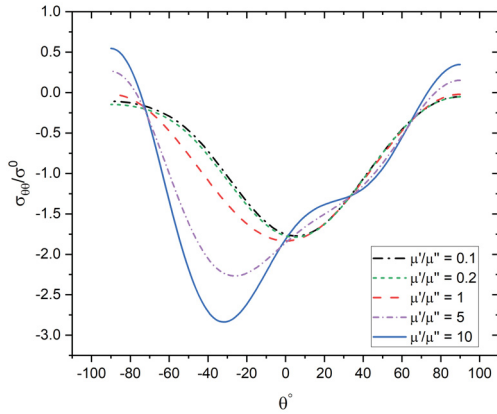


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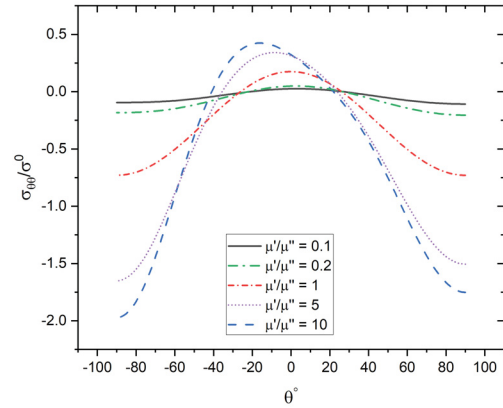


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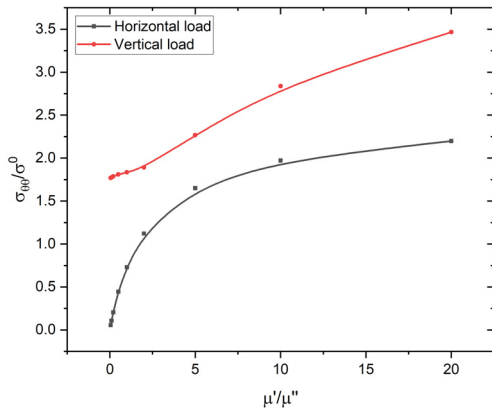
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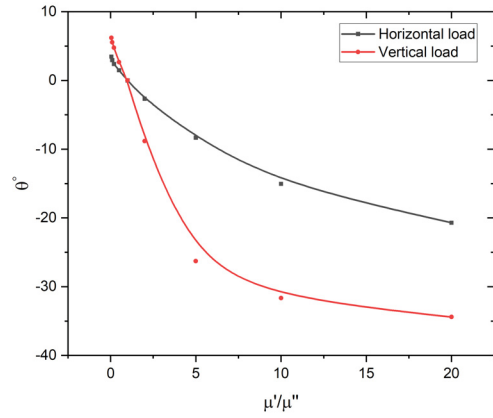
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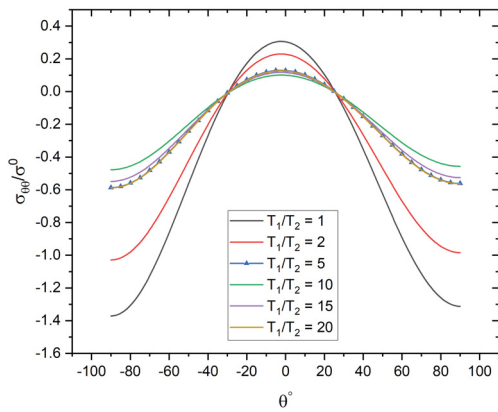


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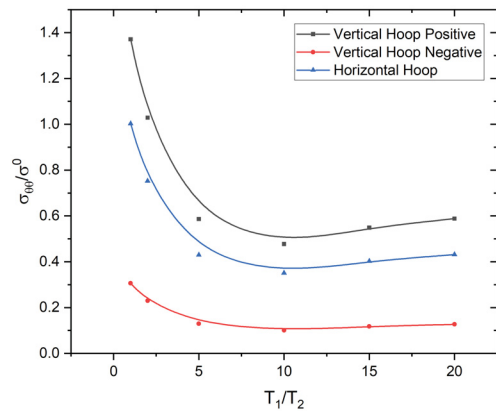


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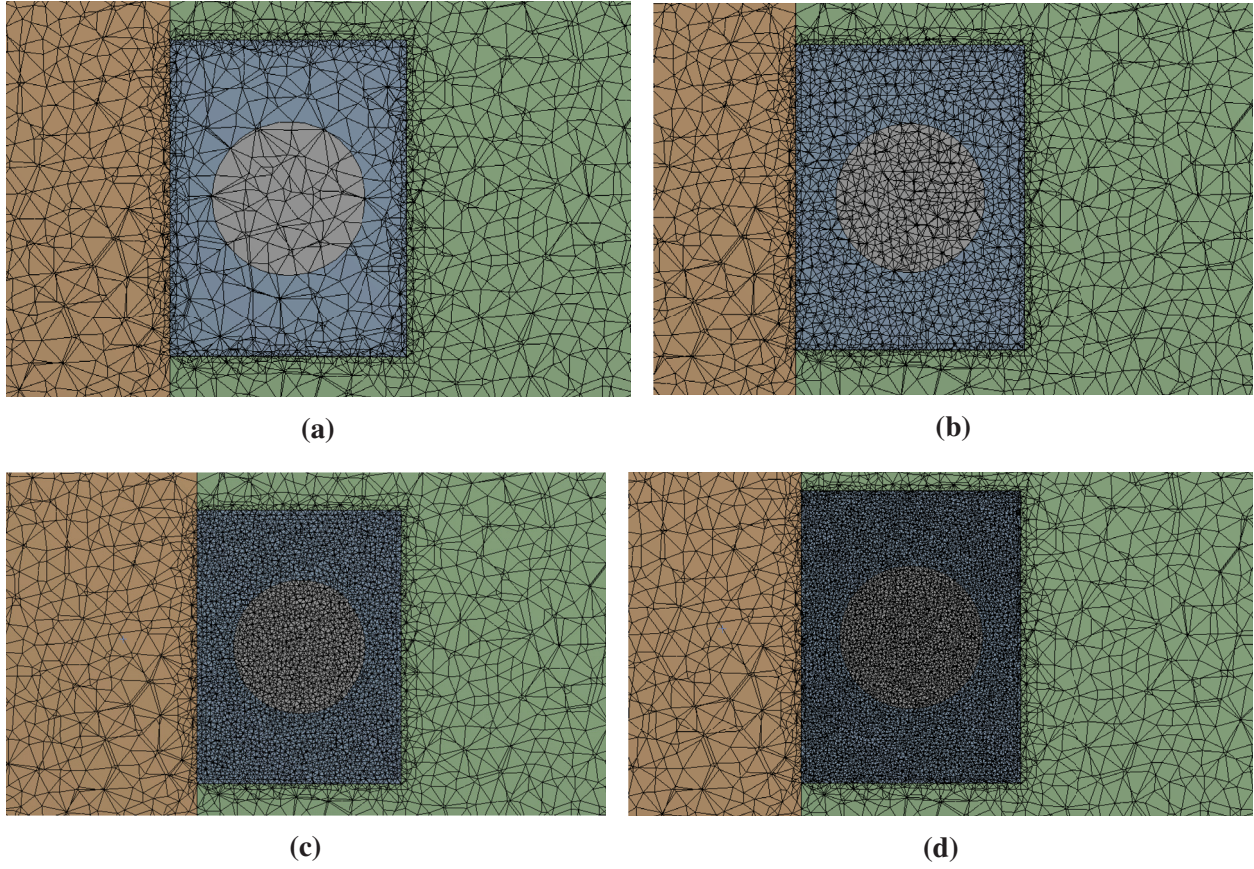


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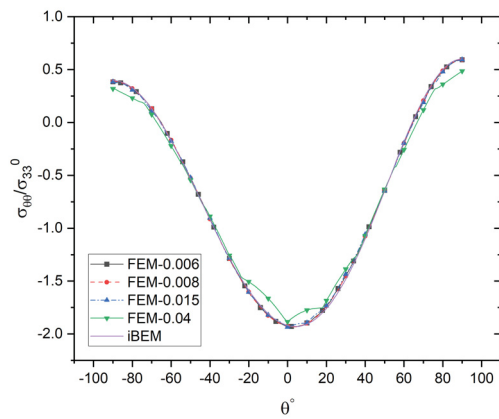


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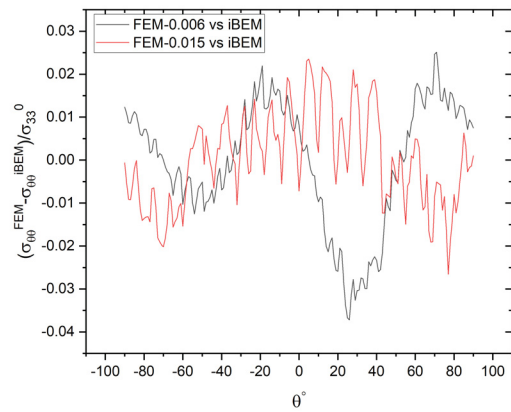
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(a)



(b)

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