

Communication

Task-Specific Phosphonium Iongels by Fast UV-Photopolymerization for Solid-State Sodium Metal Batteries

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Citation: Porcarelli, L.; Olmedo-Martínez, J.L.; Sutton, P.; Bocharova, V.; Fdz De Anastro, A.; Galceran, M.; Sokolov, A.P.; Howlett, P.C.; Forsyth, M.; Mecerreyes, D. Task-Specific Phosphonium Iongels by Fast UV-Photopolymerization for Solid-State Sodium Metal Batteries. *Gels* **2022**, *8*, 725. <https://doi.org/10.3390/gels8110725>

Academic Editors: Giao T.M. Nguyen and Jean Le Bideau

Received: 18 October 2022

Accepted: 7 November 2022

Published: 9 November 2022

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Abstract: Sodium metal batteries are an emerging technology that shows promise in terms of materials availability with respect to lithium batteries. Solid electrolytes are needed to tackle the safety issues related to sodium metal. In this work, a simple method to prepare a mechanically robust and efficient soft solid electrolyte for sodium batteries is demonstrated. A task-specific iongel electrolyte was prepared by combining in a simple process the excellent performance of sodium metal electrodes of an ionic liquid electrolyte and the mechanical properties of polymers. The iongel was synthesized by fast (<1 min) UV photopolymerization of poly(ethylene glycol) diacrylate (PEGDA) in the presence of a saturated 42%mol solution of sodium bis(fluorosulfonyl)imide (NaFSI) in trimethyl iso-butyl phosphonium bis(fluorosulfonyl)imide (P111i4FSI). The resulting soft solid electrolytes showed high ionic conductivity at room temperature ($\geq 10^{-3}$ S cm⁻¹) and tunable storage modulus (10⁴–10⁷ Pa). Iongel with the best ionic conductivity and good mechanical properties (Iongel10) showed excellent battery performance: Na/iongel/NaFePO₄ full cells delivered a high specific capacity of 140 mAh g⁻¹ at 0.1 C and 120 mAh g⁻¹ at 1 C with good capacity retention after 30 cycles.

Keywords: iongel electrolyte; polymer electrolyte; sodium metal battery

1. Introduction

Today, lithium ion batteries (LIBs) are the leading energy storage technology in the market of consumer electronics and electric mobility [1]. However, it is unlikely that LIBs alone can satisfy the demand for large-format energy storage due to the limited availability and the increasing price of lithium sources. Recent research is focusing on emerging post-lithium-ion batteries [2,3]. Multivalent ion batteries—such as magnesium, zinc, and aluminum—hold the theoretical advantage of transferring multiple charges by each ion, but the development of these technologies is still in an early stage [4]. On the other hand, sodium-ion batteries (SIBs) have gained increasing traction in academia and industry with few companies—such as Faradion (UK) and CATL (China)—near to market introduction. Sodium is a cheap and extremely abundant element that displays a very similar electrochemical behavior to lithium [5,6]. Nevertheless, SIBs still face some research challenges including lower energy densities than LIBs [4]. SIBs usually employ hard carbon anodes and carbonate-based electrolytes. Replacing hard carbon-negative electrodes with sodium metal ones could theoretically increase the energy density if suitable electrolytes for

sodium metal are found. Super-concentrated ionic liquid (IL) electrolytes have been under extensive investigation due to their superior stability as electrolytes for sodium and lithium metal batteries [7,8]. While previous studies focused on pyrrolidinium ionic liquids, only recently has there been interest in phosphonium-based ionic liquids. For instance, Hilder et al. described an electrolyte based on 42%mol NaFSI in trimethyl iso-butyl phosphonium bis(fluorosulfonyl)imide (42% mol NaFSI in P1114iFSI) electrolytes for long lasting and stable sodium metal batteries [9,10]. Despite these advantages, IL electrolytes require a porous separator and are limited by the risks of leakage [11]. To overcome these issues and enable solid-state sodium batteries, the preparation of solid gel electrolytes (also known as iongel electrolytes) has become a very popular solution. This novel class of materials combines the unique electrolyte properties of ILs with the superior mechanical properties of polymers. Ionic conductivity is one of the most important parameters in determining whether a material is a good candidate for use as an electrolyte in a battery (ionic conductivity values of the order of 10^{-4} – 10^{-3} S cm $^{-1}$ are normally required). On the other hand, the storage modulus is a measure of the mechanical properties of the material, which are reflected in the resistance of the electrolyte to dendrite growth. Along the same line, we recently demonstrated the excellent performance of iongels with sodium metal, which involves several polymer matrixes and a pyrrolidinium-type sodium ionic liquid electrolyte [12]. The goal of this work is to explore the fast UV photopolymerization method to prepare an iongel for an all-solid-state battery using a sodium-metal anode and triphylite NaFePO $_4$ as the cathode material, which includes the high-performing phosphonium ionic liquid electrolyte.

In this work, fast UV photopolymerization of poly(ethylene glycol) diacrylate (PEGDA) in the presence of 42% mol NaFSI in P1114iFSI was used to prepare self-standing iongel electrolyte membranes for application in sodium metal batteries.

2. Results and Discussion

Figure 1 shows the schematic diagram of the polymerization of cross-linked iongels. We used 2-hydroxy-2-methylpropiophenone (DAROCUR 1173) as a radical photoinitiator. We varied the amount of PEGDA cross-linker between 5 and 40 wt%, naming the iongel using the following codes: Iongel5, Iongel10, Iongel20, and Iongel40, where the number indicates the wt% of the PEGDA cross-linker. The obtained iongels were easy to handle, optically transparent, and did not leak the IL electrolyte. After polymerization, the soluble fraction was separated using a Soxhlet extractor and analyzed via $^1\text{H-NMR}$. The complete monomer conversion after the UV polymerization process was confirmed by the disappearance of the double bond signal associated with the acrylate function in the $^1\text{H-NMR}$; see Figure S1.

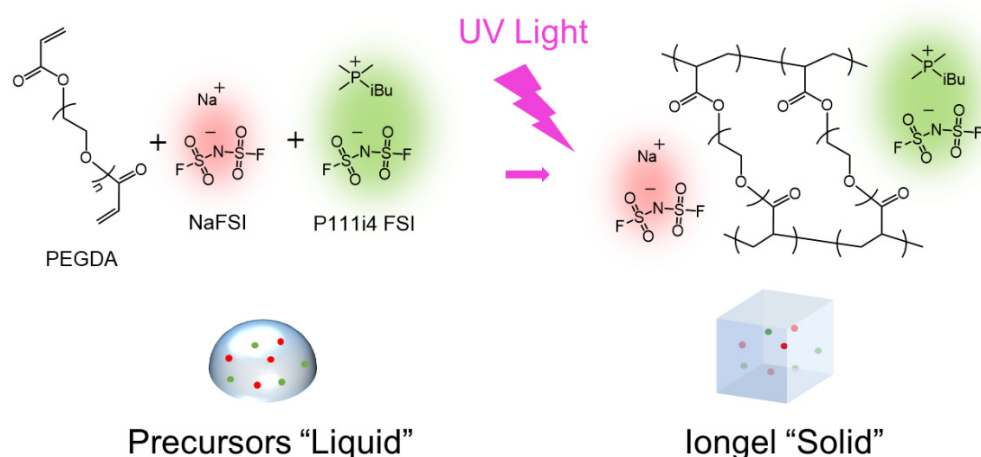


Figure 1. Synthetic representation of photopolymerization for the obtention of iongels.

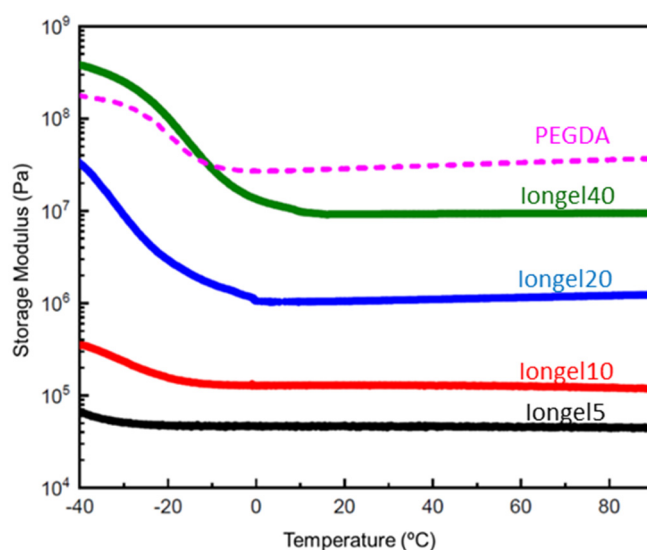
2.1. Dynamic Mechanical Analysis (DMA)

DMA was used to determine the effect of PEGDA crosslinker content on the storage modulus of the phosphonium iongel electrolytes. Figure 2 shows the storage modulus as a function of temperature (measured between -40 and 90 °C), consisting of two character-

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DMA was used to determine the effect of PEGDA crosslinker content on the storage modulus of the phosphonium iongel electrolytes. Figure 2 shows the storage modulus as a function of temperature (measured between -40 and 90 °C), consisting of two characteristic regions. In the temperature region below 0 °C, a significant modulus change was assigned to the glass transition temperature (T_g), and the T_g increased with increasing PEGDA content. It should be noted that the transition region of the measured temperature moved out of the measured temperature with PEGDA content and no estimate of T_g was possible from the DMA data. In polymers with small changes in heat capacity, it is more difficult to observe T_g by DSC, while there is a large change in the storage modulus at T_g [13].

**Figure 2.** Storage modulus as a function of temperature for the different iongels.

Above room temperature, the rubbery plateau modulus increased with the PEGDA content in the electrolyte, having a maximum value of $9 \cdot 10^6$ Pa with 40 wt% PEGDA, while the value of 100% PEGDA was $2.5 \cdot 10^7$ Pa. This is a typical increase in the rubbery modulus with increasing crosslink density. Although it was possible to obtain a gel electrolyte with 5 wt% of PEGDA, these gels were very soft.

The storage modulus value at RT for Iongel40 was higher than those reported for other solvated ionic liquids (SIL) in polymeric hosts, e.g., PEO/80% SIL (viscous liquid) [14], PEGDA/79% SIL (420 kPa) [13], or PEGDMA/80% SIL (370 kPa) [15], and these results showed stable mechanical properties within the typical operational temperature range of solid-state batteries, i.e., RT and above.

2.2. Ionic Conductivity

The ionic conductivity of the iongels was obtained from broadband dielectric spectroscopy (BDS) using the DC plateau from spectra in the conductivity representation. The IL was placed between two parallel plates made of brass and separated by a Teflon spacer ring with a thickness $L = 100$ μm . Figure 3 shows the plot of ionic conductivity between -80 and 80 °C. In general, a tradeoff was observed between ionic conductivity and the amount of PEGDA, and the sample with the lowest content of PEGDA (Iongel5) displayed the highest conductivity in the whole temperature range studied. The ionic conductivity of Iongel10 ranged between $7 \cdot 10^{-3}$ S cm^{-1} at 80 °C and $2 \cdot 10^{-8}$ S cm^{-1} at -70 °C. The conductivity decreased up to two orders of magnitude for higher PEGDA content, and the ionic conductivity of Iongel40 ranged between $2.5 \cdot 10^{-3}$ S cm^{-1} at 80 °C and $3 \cdot 10^{-11}$ S cm^{-1} at -80 °C. Interestingly, despite the significant differences of several orders of magnitude in the mechanical modulus, the conductivity only decreased by a factor of 3 in going from Iongel10 to Iongel40. Surprisingly, the pure ionic liquid electrolyte displayed a lower ionic

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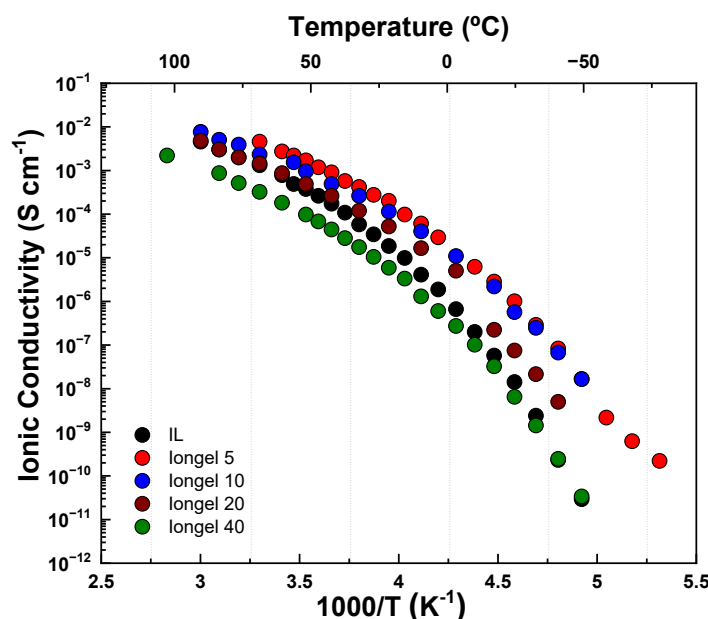


Figure 3. Ionic conductivity as a function of temperature for IL and iongels.

2.3. Battery Cell Testing

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Finally, battery cells consisting of a sodium metal anode and a NaFePO_4 cathode were assembled by discharging at 100 mA g^{-1} for 10 h under an Ar atmosphere. Figure 4a shows the charge–discharge profiles at various C-rates extracted from a cell cycled in the range of 1.5–4 V vs. Na^+/Na at 50°C . As shown, the cell displayed two charge plateaus centered on 3 V vs. Na^+/Na and a single sloping discharge profile. This asymmetric voltage range of 1.5–4 V vs. Na^+/Na at 50°C . As shown, the cell displayed two charge plateaus centered on 3 V vs. Na^+/Na and a single sloping discharge profile. This asymmetric voltage is associated with the formation of an intermediate phase during the charge and is fully consistent with previously reported results of liquid and solid-state NaFePO_4 cells [12,16–19]. At C/10, the cell delivered a maximum discharge capacity around 140 mAh g^{-1} , corresponding to 90% of the theoretical capacity of NaFePO_4 (154 mAh g^{-1}). At higher C-rates, the specific discharge capacity slightly decreased to 135 mAh g^{-1} at C/5, 130 mAh g^{-1} at C/2, and 128 mAh g^{-1} at C/1. Figure 4b shows the plot of specific capacities at various C-rates versus the cycle number and the corresponding coulombic efficiencies. During the first six formation cycles at low current rates (C/10), the coulombic efficiency increased rapidly to 99.5% and the cell delivered a stable discharge capacity. Even though the capacity at C/1 slightly decreased, the cell showed a good capacity retention and the coulombic efficiency remained close to 99.5%. Additionally, the cell recovered its initial capacity of $\approx 140 \text{ mAh g}^{-1}$ with a coulombic efficiency of around 100%. In our previous work, a cell based on a N-propyl-N-methylpyrrolidinium bis(fluorosulfonyl)imide iongel delivered a maximum discharge capacity around 145 mAh g^{-1} at C/10, corresponding to 95% of the theoretical capacity of NaFePO_4 (154 mAh g^{-1}). Despite a slightly lower initial capacity, the phosphonium cell showed a far greater capacity retention, as shown by Figure 4c, of the normalized capacity of two cells cycling at C/10. In our previous work, we observed that iongels from superconcentrated phosphonium electrolytes exhibit better battery performance compared to the previously reported pyrrolidinium counterparts due to their superior electrochemical stability [12]. The results of this work suggest that the same behavior is observed in the iongel form of these electrolytes.

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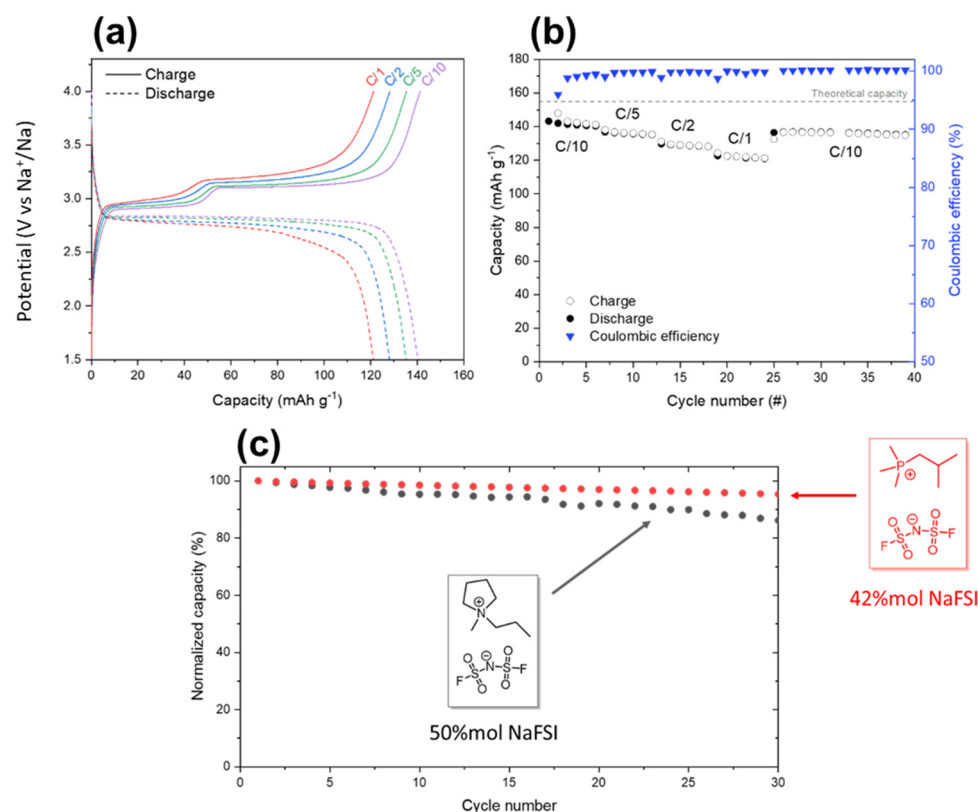


Figure 4. (a) Cycling behavior at 50 °C of a Na/iongel membrane/NaFePO₄ cycled in the range of 1.5–4 V vs. Na⁺/Na at different C-rates; (b) specific capacity vs. cycle number plot; (c) normalized capacity as a function of cycle number for the electrolyte reported in this work (red points) and electrolyte previously reported by our group (black points) [12].

3. Conclusions

Four different iongel electrolytes were prepared by UV photopolymerization using trimethyl isobutyl phosphonium bis(fluorosulfonyl)imide (P111i4FSI, Boron Molecular, Victoria, Australia) and sodium bis(fluorosulfonyl)imide (NaFSI, Solvionic, Toulouse, France) salts varying the amount of PEGDA in the liquid and solid electrolyte. The electrolytes showed a capacity retention of 100% at 50 °C for the electrolyte with 5 and 10% PEGDA in the electrolyte, ranging from 10% to 40% at 50 °C for the electrolyte with 20 and 40% PEGDA in the electrolyte. The electrolyte with 10% PEGDA showed a higher capacity retention compared to the other electrolytes. Taking into account these parameters, iongel10 was selected for testing in a sodium battery, and this electrolyte showed a higher capacity retention compared to other pyrrolidinium-based electrolytes.

4. Materials and Methods

4.1. Materials

Trimethyl isobutyl phosphonium bis(fluorosulfonyl)imide (P111i4FSI, Boron Molecular, Victoria, Australia) and sodium bis(fluorosulfonyl)imide (NaFSI, Solvionic, Toulouse, France) were dried under vacuum at 50 °C and transferred inside an Ar-filled glove box before use. Poly(ethylene glycol) diacrylate M_n 575 (PEGDA; Sigma-Aldrich, Madrid, Spain) was passed through a basic alumina column to remove the hydroquinone monomethyl ether inhibitor (MEHQ), filtered with a 0.45 µm syringe filter, and kept refrigerated at 5 °C before use. 2-hydroxy-2-methylpropiophenone (DAROCUR 1173, Sigma-Aldrich) was used as received.

4.2. Sample Preparation

A saturated 42% mol electrolyte solution of NaFSI in P111i4FSI was prepared inside an Ar-filled glovebox by stirring the solution on a hot plate at 50 °C, and stored inside the glovebox until use. A 2.5% wt monomer solution of DAROCUR 1173 in PEGDA was prepared outside the glovebox before use. Ionogel membranes were prepared by mixing different weight amounts of the electrolyte and monomer solution (Table S1). The mixtures were cast on a silicone mold irradiated with a UV lamp for 90 s twice. The iongel membranes were kept for 24 h under vacuum at 90 °C and stored in an argon-filled glovebox until use. The membranes were circular disks (diameter = 14 mm; average thickness = 250 µm).

4.3. Physical–Chemical Characterization

DMA experiments were performed on a PerkinElmer DMA 8000 in tension mode with a heating rate of 5 °C min^{−1}, at a 1 Hz frequency and strain of 25 µm, and in a N₂ atmosphere. Broadband dielectric spectra in the frequency range of 10^{−1} to 10⁶ Hz were measured using a Novocontrol Concept-80 system, which includes an Alpha-A impedance analyzer and a Quatro Cryosystem temperature control unit. The samples were placed between the stainless-steel parallel plates with a 20 mm diameter, and the separation between the electrodes was determined by the film thickness, approximately 0.2 mm. The samples were placed inside the cryostat in a dry nitrogen atmosphere. The samples were equilibrated for at least 15 min after each temperature step to achieve thermal stabilization within 0.2 K.

The *Triphylite*-NaFePO₄ cathode active material was synthesized using a two-step reaction reported previously [20].

4.4. Cell Assembly and Testing

Na/iongel/NaFePO₄ cells were assembled for testing in sodium batteries. Sodium metal was used as the anode. The electrolytes and the electrodes were placed between two stainless-steel spacers (o.d. = 16 mm; thickness = 0.5 mm), and the cells were prepared inside the glovebox in argon atmosphere. These cells were measured in a VMP3 Biologic potentiostat at 50 °C and cycled at C/20 cycled in a potential range of 1.5–4 V for 30 cycles.

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/gels8110725/s1>, Figure S1: H¹-NMR spectra of soluble fraction after Soxhlet extraction; Table S1: Composition of the different electrolytes prepared.

Author Contributions: The manuscript was written through contributions of all authors. Conceptualization: D.M. and M.F.; investigation: L.P., P.S., V.B., A.F.D.A., M.G., A.P.S. and P.C.H.; data curation: J.L.O.-M. and L.P.; supervision: D.M. and M.F.; validation: M.F., L.P. and D.M. writing—original draft: J.L.O.-M. and L.P.; writing—review and editing: J.L.O.-M. and D.M. All authors have read and agreed to the published version of the manuscript.

Funding: L.P. has received funding from the European Union’s Horizon 2020 research and innovation programme under the Marie Skłodowska–Curie grant agreement No. 797295. P.S. has been funded by the SNSF (Swiss National Science Foundation) under project number P2FRP2_191846. D.M., M.G., and A.F.A. acknowledge the funding by the Basque Government through Elkartek KK-2020/00078 and Agencia Estatal de Investigación (PLEC2021-007929). D.M. and M.G. also acknowledge the funding by Agencia Estatal de Investigación (PID2020-119026GB-I00 and PID2019-107468RB-C22), respectively. A.P.S. acknowledges financial support for dielectric measurements and data discussions by NSF (award CHE-2102425).

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: Not applicable.

Acknowledgments: V.B. acknowledges support from the Laboratory Directed Research and Development Program of Oak Ridge National Laboratory (ORNL), managed by UT-Battelle, LLC for the U.S. Department of Energy under Contract No. DE-AC05-00OR22725.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Sun, C.; Liu, J.; Gong, Y.; Wilkinson, D.P.; Zhang, J. Recent advances in all-solid-state rechargeable lithium batteries. *Nano Energy* **2017**, *33*, 363–386. [[CrossRef](#)]
2. Choi, J.W.; Aurbach, D. Promise and reality of post-lithium-ion batteries with high energy densities. *Nat. Rev. Mater.* **2016**, *1*, 16013. [[CrossRef](#)]
3. Kulova, T.L.; Fateev, V.N.; Seregina, E.A.; Grigoriev, A.S. A brief review of post-lithium-ion batteries. *Int. J. Electrochem. Sci.* **2020**, *15*, 7242–7259. [[CrossRef](#)]
4. Ponrouch, A.; Bitenc, J.; Dominko, R.; Lindahl, N.; Johansson, P.; Palacín, M.R. Multivalent rechargeable batteries. *Energy Storage Mater.* **2019**, *20*, 253–262. [[CrossRef](#)]
5. Yang, J.; Zhang, H.; Zhou, Q.; Qu, H.; Dong, T.; Zhang, M.; Tang, B.; Zhang, J.; Cui, G. Safety-enhanced polymer electrolytes for sodium batteries: Recent progress and perspectives. *ACS Appl. Mater. Interfaces* **2019**, *11*, 17109–17127. [[CrossRef](#)] [[PubMed](#)]
6. Gebert, F.; Knott, J.; Gorkin III, R.; Chou, S.-L.; Dou, S.-X. Polymer electrolytes for sodium-ion batteries. *Energy Storage Mater.* **2021**, *36*, 10–30. [[CrossRef](#)]
7. Rakov, D.A.; Chen, F.; Ferdousi, S.A.; Li, H.; Pathirana, T.; Simonov, A.N.; Howlett, P.C.; Atkin, R.; Forsyth, M. Engineering high-energy-density sodium battery anodes for improved cycling with superconcentrated ionic-liquid electrolytes. *Nat. Mater.* **2020**, *19*, 1096–1101. [[CrossRef](#)] [[PubMed](#)]
8. Sun, J.; O'Dell, L.A.; Armand, M.; Howlett, P.C.; Forsyth, M. Anion-Derived Solid-Electrolyte Interphase Enables Long Life Na-Ion Batteries Using Superconcentrated Ionic Liquid Electrolytes. *ACS Energy Lett.* **2021**, *6*, 2481–2490. [[CrossRef](#)]
9. Hilder, M.; Howlett, P.C.; Saurel, D.; Gonzalo, E.; Armand, M.; Rojo, T.; Macfarlane, D.R.; Forsyth, M. Small quaternary alkyl phosphonium bis (fluorosulfonyl) imide ionic liquid electrolytes for sodium-ion batteries with P2-and O3-Na_{2/3} [Fe_{2/3}Mn_{1/3}] O₂ cathode material. *J. Power Sources* **2017**, *349*, 45–51. [[CrossRef](#)]
10. Ferdousi, S.A.; O'Dell, L.A.; Sun, J.; Hora, Y.; Forsyth, M.; Howlett, P.C. High-Performance Cycling of Na Metal Anodes in Phosphonium and Pyrrolidinium Fluoro (sulfonyl) imide Based Ionic Liquid Electrolytes. *ACS Appl. Mater. Interfaces* **2022**, *14*, 15784–15798. [[CrossRef](#)] [[PubMed](#)]
11. Kalhoff, J.; Eshetu, G.G.; Bresser, D.; Passerini, S. Safer Electrolytes for Lithium-Ion Batteries: State of the Art and Perspectives. *ChemSusChem* **2015**, *8*, 2154–2175. [[CrossRef](#)] [[PubMed](#)]
12. Fdz De Anastro, A.; Porcarelli, L.; Hilder, M.; Berlanga, C.; Galceran, M.; Howlett, P.; Forsyth, M.; Mecerreyes, D. UV-cross-linked ionogels for all-solid-state rechargeable sodium batteries. *ACS Appl. Energy Mater.* **2019**, *2*, 6960–6966. [[CrossRef](#)]
13. Chereddy, S.; Aguirre, J.; Dikin, D.; Wunder, S.L.; Chinnam, P.R. Gel electrolyte comprising solvate ionic liquid and methyl cellulose. *ACS Appl. Energy Mater.* **2019**, *3*, 279–289. [[CrossRef](#)]
14. Kido, R.; Ueno, K.; Iwata, K.; Kitazawa, Y.; Imaizumi, S.; Mandai, T.; Dokko, K.; Watanabe, M. Li⁺ ion transport in polymer electrolytes based on a glyme-Li salt solvate ionic liquid. *Electrochim. Acta* **2015**, *175*, 5–12. [[CrossRef](#)]
15. Hubble, D.; Qin, J.; Lin, F.; Murphy, I.A.; Jang, S.-H.; Yang, J.; Jen, A.K.-Y. Designing solvate ionogel electrolytes with very high room-temperature conductivity and lithium transference number. *J. Mater. Chem. A* **2018**, *6*, 24100–24106. [[CrossRef](#)]
16. Casas-Cabanas, M.; Roddatis, V.V.; Saurel, D.; Kubiak, P.; Carretero-González, J.; Palomares, V.; Serras, P.; Rojo, T. Crystal chemistry of Na insertion/deinsertion in FePO₄–NaFePO₄. *J. Mater. Chem.* **2012**, *22*, 17421–17423. [[CrossRef](#)]
17. Wongittharom, N.; Lee, T.-C.; Wang, C.-H.; Wang, Y.-C.; Chang, J.-K. Electrochemical performance of Na/NaFePO₄ sodium-ion batteries with ionic liquid electrolytes. *J. Mater. Chem. A* **2014**, *2*, 5655–5661. [[CrossRef](#)]
18. Galceran, M.; Saurel, D.; Acebedo, B.; Roddatis, V.V.; Martin, E.; Rojo, T.; Casas-Cabanas, M. The mechanism of NaFePO₄ (de) sodiation determined by in situ X-ray diffraction. *Phys. Chem. Chem. Phys.* **2014**, *16*, 8837–8842. [[CrossRef](#)] [[PubMed](#)]
19. Berlanga, C.; Monterrubio, I.; Armand, M.; Rojo, T.; Galceran, M.; Casas-Cabanas, M. Cost-effective synthesis of triphylite-NaFePO₄ cathode: A zero-waste process. *ACS Sustain. Chem. Eng.* **2019**, *8*, 725–730. [[CrossRef](#)]
20. Saurel, D.; Giner, M.; Galceran, M.; Rodríguez-Carvajal, J.; Reynaud, M.; Casas-Cabanas, M. The triphylite NaFe_{1-y}MnyPO₄ solid solution (0 ≤ y ≤ 1): Kinetic strain accommodation in Na_xFe_{0.8}Mn_{0.2}PO₄. *Electrochim. Acta* **2022**, *425*, 140650. [[CrossRef](#)]