

## Designing dual-functional metal-organic frameworks for photocatalysis

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**Abstract**

Hydrogen (H<sub>2</sub>) is an ideal alternative to fossil fuels as it is sustainable and environmentally friendly. Hydrogen production using semiconductor-based materials has been extensively investigated; most studies, however, rely on the use of sacrificial electron donors to consume the photogenerated holes, which wastes their oxidizing potential. Dual-functional photocatalysis (DFP) couples the production of H<sub>2</sub> with the oxidation of organic molecules, enabling simultaneous utilization of both photogenerated species. To develop efficient materials for DFP, herein, we investigate the interplay of electron/hole dynamics and photophysical properties of metal-organic frameworks (MOFs) using experimental and computational techniques. Four zirconium-based MOFs (UiO-66 analogs) were synthesized using different nitrogen-functionalized ligands. We used benzenethiol in place of a sacrificial reagent to enable simultaneous H<sub>2</sub> production and benzenethiol oxidation to sulfide-based products. We demonstrated that Pt/UiO-66-pz (Pt: platinum nanoparticles, pz: pyrazine) is the most efficient dual-functional photocatalyst as it achieved the highest H<sub>2</sub> production rates and second-best benzenethiol conversion. Our results shed light on the complex DFP process, wherein the interplay of light absorption, conductivity, band alignment, and charge separation and transfer capabilities are vital for enhancing the dual-functional photocatalytic activity of MOFs.

## Introduction

The production of hydrogen fuel ( $H_2$ ) using two abundant resources, water, and solar energy, is considered a clean and sustainable method for green fuel.<sup>1-4</sup> Many semiconductor-based materials have been tested as photocatalysts for water-splitting to generate  $H_2$ . Upon irradiation of a photocatalyst, electron/hole pairs (excitons) are generated, each able to participate in redox reactions. While the photogenerated electrons promote the  $H_2$  evolution reaction (HER), the oxygen ( $O_2$ ) evolution reaction (OER) can utilize holes to produce  $O_2$ .<sup>5-6</sup> Overall water splitting involves the production of both  $H_2$  and  $O_2$ , but it can suffer from slow kinetics due to the  $O_2$  half-reaction, which requires at least four holes, compared to the HER, which only requires two electrons. To overcome this challenge, sacrificial electron donors such as triethylamine (TEA) and triethanolamine (TEOA) are used to suppress the OER and accelerate the HER.<sup>7</sup> Although this strategy can lead to high  $H_2$  evolution rates (exceeding  $2000 \mu\text{mol h}^{-1} \text{g}^{-1}$ ),<sup>8</sup> sacrificial electron donors are themselves valuable products, and the  $H_2$  generated from their oxidation does not compensate for the energy required to produce them.<sup>9-10</sup> Therefore, it is desirable to take advantage of the oxidizing nature of the photogenerated holes to break down pollutants or produce valuable end products. A promising solution is designing a dual-functional photoredox system that can simultaneously utilize both the electrons and holes.

Dual-functional photocatalysis (DFP) couples the production of  $H_2$  with the oxidation of organic molecules (i.e., organic synthesis), enabling the utilization of both photogenerated species (Figure 1).<sup>9, 11-13</sup> For instance,  $TiO_2$  has been applied in concurrent  $H_2$  production ( $180 \mu\text{mol h}^{-1} \text{g}^{-1}$ ) and dehydrogenative lactonization of 1,2-benzenedimethanol, with a phthalide yield of 90%.<sup>14</sup> Although some progress has been made in understanding the underlying mechanisms of DFP, it is still unclear how to

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3 maximize the output of both half-reactions simultaneously.<sup>12</sup> Without guidance on  
4 tuning the activity and selectivity of semiconductors for DFP, most of the studies  
5 reported are based on a trial-and-error approach.<sup>11-12</sup> Therefore, assessing the  
6 fundamental processes in DFP becomes crucial in designing efficient dual-functional  
7 photocatalysts.  
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15 After generating the electron/hole pairs, it is vital to transfer these charge carriers to the  
16 reactants (water and an organic molecule) with minimal energy loss. Energy loss can  
17 be due to charge recombination, the formation of a deep trap state, or multiple  
18 intermediate energy dissipation steps within or between different excited states.  
19 Femtosecond transient absorption (fs-TA) spectroscopy can be used to gain insights  
20 into how the photogenerated charge carrier dynamics affect the overall charge transfer  
21 and energy usage.<sup>15-17</sup> As an ultrafast electronic spectroscopic technique, fs-TA  
22 employs a pump pulse and a time-delayed probe pulse to track the excited state  
23 dynamics with characteristic time constants. For example, the solvation-aided excited  
24 state stabilization could improve charge transfer efficiency, in accord with a blue-  
25 shifted excited-state absorption (ESA) peak.<sup>18-19</sup>  
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41 Notably, trap states within a material can also affect the charge carrier dynamics. A  
42 shallow trap allows electrons to be rapidly transferred to the reactant or surrounding  
43 solvent, affording high reaction rates. In contrast, a much longer lifetime may indicate  
44 the recombination of deeply trapped carriers, leading to energy loss and reduced charge  
45 transfer. Therefore, materials that possess fewer intermediate energy transfer steps and  
46 high charge transfer rates (e.g., with accessible charge carriers out of a shallow trap  
47 state) would be beneficial for efficient DFP.<sup>19-23</sup>  
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3 Meanwhile, the energies of both the valence band (VB) and conduction band (CB) of  
4 the photocatalyst, and their alignment/overlap with the highest occupied molecular  
5 orbital (HOMO) of the organic molecule, are critical for achieving thermodynamically  
6 favorable DFP reactions (Figure 1). For water reduction, the CB energy must be more  
7 negative than the reduction potential of  $H^+/H_2$  (0.0 eV vs. Normal Hydrogen Electrode  
8 (NHE) at pH=0). For the oxidation of the organic molecule, the HOMO of the  
9 molecules needs to be less positive than the VB of the photocatalyst. Hence, proper  
10 alignment of the VB and CB of the photocatalyst with the HOMO of the organic  
11 molecule and the  $H_2$  evolution potential is crucial in DFP to achieve high  $H_2$  evolution  
12 and oxidation rates.<sup>24</sup>

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27 The synergy between light absorption, charge dynamics, and band alignment must be  
28 considered for designing suitable DFP materials. Metal-organic frameworks (MOFs)  
29 with versatile structures and tunable physicochemical properties present an ideal  
30 platform for such a task. As photocatalytic materials, MOFs are theoretically superior  
31 to traditional semiconductors due to their: (i) high porosity, which allows co-catalysts  
32 and organic molecules to diffuse within them, increasing their interactions<sup>22</sup>, and (ii)  
33 highly tunable structures that allow their optoelectronic properties, conductivity, and  
34 hydrolytic stability to be modified by changing the metal ions or functionalizing the  
35 organic ligand.<sup>25-30</sup> Moreover, traditional semiconductors are limited by their ability to  
36 primarily absorb UV light, whereas MOFs can be tuned to absorb visible light that  
37 represents a more significant fraction of solar energy reaching Earth. Hence,  
38 photocatalysts with a broader light absorption range can be useful in DFP.<sup>26, 31</sup>

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MOFs have been previously applied in DFP, such as MIL-125-NH<sub>2</sub>, which  
photodegraded Rhodamine B while producing  $H_2$  at a rate of  $335 \mu\text{mol h}^{-1} \text{g}^{-1}$ .<sup>24</sup> PCN-  
777 simultaneously generated  $H_2$  ( $30 \mu\text{mol h}^{-1} \text{g}^{-1}$ ) and oxidized benzylamine with 90%

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3 selectivity.<sup>32</sup> Other examples of DFP include oxidizing water while reducing carbon  
4 dioxide (CO<sub>2</sub>) or dinitrogen (N<sub>2</sub>) to methanol (CH<sub>4</sub>), or ammonia (NH<sub>3</sub>), respectively.<sup>33-</sup>  
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8 <sup>34</sup> The aforementioned reactions involve gases, making it challenging to determine their  
9 specific interactions with the MOF photocatalyst. We envision a DFP system where  
10 organic molecules can diffuse within the pore space and interact with localized holes  
11 on the organic linkers while the delocalized, photogenerated electrons can reduce water  
12 to H<sub>2</sub>. Detailed studies on the underlying mechanisms are lacking, and the essential  
13 photophysical properties for MOFs to maximize DFP remain elusive.  
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22 In this work, we describe for the first time the impact of light absorption, charge carrier  
23 dynamics, and band alignment in MOFs for DFP. The fs-TA spectroscopy, computed  
24 Projected Density of States (PDOS), and computational predictions of bandgap  
25 alignments based on Density-Functional Theory (DFT) were used in concert to  
26 investigate the role of photogenerated charges within the MOF photocatalyst.<sup>32, 35</sup> Four  
27 zirconium-based MOFs (UiO-66 analogs) were synthesized using the functionalized  
28 terephthalic acid ligands: aminoterephthalic acid, 2,5-pyridinedicarboxylic acid (pd),  
29 and 2,5-pyrazinedicarboxylic acid (pz). Compared to terephthalic acid used in UiO-66,  
30 the pd and pz ligands have nitrogen (N) atoms in the aromatic ring. Each N-atom  
31 possesses a lone pair of electrons, resulting in higher electron density, enhanced  
32 electron delocalization, and conductivity, thereby benefitting DFP in general. The  
33 incorporation of N-atoms attached to or within the benzene ring alters the light  
34 absorption, bandgap, and charge transfer properties of each MOF. The UiO-66 analogs  
35 were employed in DFP, coupling water reduction for H<sub>2</sub> production with the oxidation  
36 of benzenethiol. The relationship between the photophysical properties of each MOF  
37 and their catalytic activity as dual-functional photocatalysts was elucidated.  
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## Experimental Section

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3 **Materials.** All the chemicals and solvents in this work were purchased from  
4 commercial sources and used without further purification.  
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8 **Synthesis of UiO-66.** A mixture of zirconium(IV) chloride (20 mg, 0.08 mmol) and  
9 benzoic acid (500 mg, 0.6 mmol) was dissolved in 5 mL dimethylformamide (DMF),  
10 followed by the addition and dissolution of terephthalic acid (11.4 mg, 0.06 mmol). The  
11 mixture was stirred to form a homogeneous solution, heated in a Pyrex vial at 120 °C  
12 for 72 h, and cooled to room temperature. The as-obtained products were filtered and  
13 washed with DMF (3×). Finally, the as-prepared UiO-66 powder was air-dried  
14 overnight.  
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25 **Synthesis of UiO-66-NH<sub>2</sub>.** UiO-66-NH<sub>2</sub> was synthesized in a similar manner to UiO-  
26 66, except that 2-aminoterephthalic acid (12.4 mg, 0.06 mmol) was used instead of  
27 terephthalic acid.  
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33 **Synthesis of UiO-66-pd.** A mixture of zirconium(IV) chloride (143.4 mg, 0.6 mmol)  
34 and 2,5-pyridinedicarboxylic acid (73.3 mg, 0.4 mmol) was dissolved in 4.5 mL formic  
35 acid and 0.5 mL deionized (DI)-water. The mixture was treated with ultrasound for 20  
36 minutes to form a homogeneous solution, then heated in a Pyrex vial at 120 °C for 3 h  
37 and cooled to room temperature. The suspension was separated by centrifugation and  
38 washed with DI-water and ethanol. The product was dried under ambient conditions.  
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48 **Synthesis of UiO-66-pz.** UiO-66-pz was synthesized in a similar manner to UiO-66-  
49 pd, except that 2,5-pyrazinedicarboxylic acid (89.5 mg, 0.4 mmol) was used instead of  
50 2,5-pyridinedicarboxylic acid.  
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## 58 **Results and Discussion**

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3 **Synthesis and Characterization.** The UiO-66 ( $Zr_6O_4(OH)_4(BDC)_6$ , where BDC=  
4 terephthalic acid) family of MOFs was selected in this study due to their thermal and  
5 hydrolytic stability. UiO-66 analogs can be synthesized using different dicarboxylate  
6 ligands while retaining structural morphology<sup>36</sup>; we synthesized UiO-66 analogs using  
7 three different N-functionalized ligands: 2-aminoterephthalic acid (UiO-66-NH<sub>2</sub>), 2,5-  
8 pyridinedicarboxylic acid (pd) (UiO-66-pd), and 2,5-pyrazinedicarboxylic acid (pz)  
9 (UiO-66-pz) (Figure 2a). The 2-aminoterephthalic acid has a –NH<sub>2</sub> group attached to  
10 the aromatic ring. The pd and pz ligands have one and two N-atoms in the within the  
11 aromatic ring, respectively. Each functionalized MOF has the same cubic structure as  
12 UiO-66, crystallizing in the space group *Fm-3m* and possessing **fcu** topology.<sup>37</sup> The  
13 secondary building units (SBUs) of each MOF are composed of cuboctahedral  
14 zirconium-oxide clusters with the formula  $Zr_6O_4(OH)_4$ , and each SBU is 12-connected  
15 to the functionalized terephthalic acid linkers through their carboxylate groups.<sup>37-38</sup>  
16 Powder X-ray diffraction (PXRD) patterns confirmed that these MOFs are crystalline,  
17 isostructural, and phase pure (Figure 2b). Nitrogen (N<sub>2</sub>) isotherms collected at 77 K and  
18 1 bar revealed that all MOFs are microporous with Brunauer–Emmett–Teller (BET)  
19 surface areas of 734, 705, 678, and 679 m<sup>2</sup>/g for UiO-66, UiO-66-NH<sub>2</sub>, UiO-66-pd, and  
20 UiO-66-pz, respectively (Figure 2c). Scanning electron microscopy (SEM) images  
21 showed that all the MOFs have similar octahedral crystal morphologies (Figure S1).  
22 Thermogravimetric analysis (TGA) showed that all UiO-66 analogs are stable up to 300  
23 °C. The mass loss of all the MOF before 300 °C was correlated to the removal of guest  
24 solvent molecules residing within the pores of the MOFs. The mass loss of all the MOF  
25 at 300 °C or above this temperature is due to the structure collapsing of the MOF  
26 (Figure S2). Electrochemical Impedance Spectroscopy (EIS) was used to determine the  
27 electrical conductivity of each MOF. A Nyquist plot designates low charge-transfer  
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3 resistance when the radius of the semi-circle formed is small<sup>20, 39-40</sup>, meaning that the  
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5 charge carriers can migrate more freely through the MOF and participate in redox  
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7 reactions. In the Nyquist plot (Figure 2d), UiO-66-pz has the lowest charge-transfer  
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9 resistance, followed by UiO-66-pd, UiO-66-NH<sub>2</sub>, and UiO-66. The EIS data reveal that  
10  
11 the additional electron lone pair(s) provided by N atoms in the aromatic ring increases  
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13 the MOF conductivity.  
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18 **Photophysical Properties.** Ultraviolet-visible (UV-vis) spectroscopy and calculated  
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20 PDOS were utilized to characterize the light-harvesting properties of each MOF. UiO-  
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22 66-NH<sub>2</sub> shows an absorption band up to 430 nm, attributable to the amino-  
23  
24 functionalized ligand acting as a photosensitizer.<sup>41</sup> In contrast to the similar absorption  
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26 edges (335 nm) of UiO-66 and UiO-66-pd with one N-atom in the aromatic ring (Figure  
27  
28 2e), the notably extended absorption edge of UiO-66-pz to 380 nm implies that two N-  
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30 atoms in the aromatic ring affects the light-harvesting capabilities. The enhanced  
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32 electron-donating properties of the N-heterocycle in UiO-66-pz red-shift the absorption  
33  
34 from UiO-66 and UiO-66-pd. The bandgap for each MOF was calculated from the  
35  
36 diffuse reflectance spectroscopy data<sup>42</sup>, with UiO-66-NH<sub>2</sub> (2.88 eV) and UiO-66-pz  
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38 (3.24 eV) having smaller band gaps than UiO-66 (3.84 eV) and UiO-66-pd (3.89 eV)  
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40 (Figure 2e, Figure S3). The smaller bandgap of UiO-66-NH<sub>2</sub> and UiO-66-pz implies  
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42 that less energy is required to generate charge carriers when compared to UiO-66 and  
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44 UiO-66-pd.  
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51 The computed PDOS revealed that the N *p*-orbitals in UiO-66-pd and UiO-66-pz and  
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53 the C *p*-orbitals in UiO-66-NH<sub>2</sub> contribute most to the states close to the valence band  
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55 maximum (Figure S4). The Ultra p X-ray photoelectron spectroscopy (XPS) revealed  
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57 that the energy levels of the valence bands for UiO-66-pd and UiO-66-pz are 2.39 and  
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59 1.79 eV respectively (Figure S5). The valence states close to the band gap are mostly  
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3 ligand in character, dominated by the N and C  $p$ -orbitals, whereas the conduction bands  
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5 have a mixture of ligand N( $p$ ) and C( $p$ ) and Zr  $d$ -orbitals. Calculations reveal that the  
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7 substitutions in the N-containing MOFs have the effect of pushing occupied localized  
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9 ligand states between the Highest Occupied Crystal Orbital (HOCO) and Lowest  
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11 Unoccupied Crystal Orbital (LUCO) of UiO-66, which narrows their respective band  
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15 gaps.

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18 The localization of the electrons and holes in each MOF structure was determined using  
19  
20 the DFT-calculated isodensity surfaces of the HOCO and LUCO for the hole and  
21  
22 electron, respectively (Figure 3). The photogenerated electrons in all four MOFs were  
23  
24 primarily located on the ligand. The holes of UiO-66-NH<sub>2</sub>, UiO-66-pd, and UiO-66-pz  
25  
26 are mainly situated on the ligand of each MOF. In contrast, the holes of UiO-66 are  
27  
28 spatially positioned on the nodal and linker O atoms, which differs from the location of  
29  
30 the electrons, thus leading to lower exciton binding energy and better hole mobility (but  
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32 reduced charge transfer capabilities, see fs-TA results below). Notably, the N-  
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34 functionalized ligands seem to exert a more significant impact on the location of holes  
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36 than that of the electrons.  
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42 Fs-TA experiments were performed on all four MOFs to elucidate the excited-state  
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44 electronic dynamics governing the subsequent redox reactions (Figure 4).<sup>19</sup> All four  
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46 MOFs exhibit a broad ESA band: UiO-66 and UiO-66-NH<sub>2</sub> show a narrow band  
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48 centered at ~598 and 630 nm, respectively; UiO-66-pd shows a bluer broad band  
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50 centered at ~520 nm, while UiO-66-pz displays two maxima at 515 and 575 nm. Since  
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52 the TA kinetics are generally affected by spectral overlap, global analysis was  
53  
54 performed to retrieve the characteristic time constants of the excited state decay, which  
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56 can be best fit using four components. Under UV excitation, the solvent (acetonitrile)  
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58 shows a weak TA signal that decays completely within 1 ps; therefore, we can assign  
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3 the first time constant ( $\tau_1$ ) to Franck–Condon relaxation of the MOFs convoluted with  
4  
5 the solvent response.  
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9 The  $\tau_2$  component exhibits a clear blueshift of the ESA peak of UiO-66-pd and UiO-  
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11 66-pz, whereas in UiO-66-NH<sub>2</sub>, there is a much smaller frequency shift and primarily  
12  
13 an intensity change. Interestingly, UiO-66 shows a small redshift on this timescale,  
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15 which may be due to a rapidly decaying blue shoulder of the ESA band that is broader  
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17 (more heterogeneous) than that in UiO-66-NH<sub>2</sub>. As depicted on the excited-state  
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19 potential energy surface (Figure 4a),  $\tau_2$  corresponds to ultrafast stabilization of the  
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21 excited state by solvation (i.e., reorientation of solvent molecules afforded by the  
22  
23 MOFs' porosity), which may facilitate the exciton dissociation and charge transfer from  
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25 the ligand to a trap state. The pertinent time constant of  $\sim 2\text{--}5$  ps is larger than the  $\sim 1$  ps  
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27 exciton dissociation time observed in perovskites such as FAPbI<sub>3</sub><sup>43</sup>, consistent with a  
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29 higher exciton binding energy in these MOFs. Moreover, due to different structural  
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31 defects, the intrinsically heterogeneous MOF structure could provide trap states that  
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33 prime the system for separating and transferring the photogenerated charge carriers.<sup>20</sup>  
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35 A similar transition has been observed for zirconium-based MOFs with naphthalene  
36  
37 dicarboxylate ligands, where the transition was assigned to a ligand-to-cluster charge  
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39 transfer mechanism with the formation of a charge-separated state.<sup>19, 44</sup> Regardless of  
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41 the exact nature of the product state, an ultrafast energy stabilization is achieved on the  
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43 few ps timescale so that charge carriers emerge mainly in association with a shallow  
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45 trap state.  
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53 Notably,  $\tau_3$  and  $\tau_4$  represent the majority of ESA intensity decay (80–90%) and are most  
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55 relevant to the redox performance of UiO-66 analogs. In general, a long-lived excited  
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57 state is beneficial for DFP with more time for the excited state potential to be harvested  
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3 for catalysis. However, the longest lifetime of  $\tau_4$  shows only a slight variation between  
4 samples from  $\sim 690$  ps (UiO-66-NH<sub>2</sub>) to 900 ps (UiO-66), which cannot be the main  
5  
6 factor explaining the performance differences between these materials. We can attribute  
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8 it to the recombination of trapped carriers from deep trap states that are less accessible  
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10 for reduction (Figure 4a).<sup>20, 45</sup> However, even though the deep-trapped electrons may  
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12 not be readily available for reduction reactions, the corresponding holes could become  
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14 available for oxidation reactions, and the largest  $\tau_4$  in UiO-66 is in accord with its  
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16 greater hole mobility and the best oxidation potential among all the MOFs tested (see  
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18 below).

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25 In contrast,  $\tau_3$  shows a clear change between UiO-66/UiO-66-NH<sub>2</sub> ( $\sim 55$  ps) and UiO-  
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27 66-pd/UiO-66-pz ( $\sim 100$  ps). This notable and consistent difference indicates that the  
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29 different electronic landscape with the pd and pz ligands imposes a slower relaxation  
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31 step on the ultrafast timescale, which is likely associated with a shallow trap state of  
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33 MOF that plays an important role in charge transfer. In other words, the prolonged  
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35 lifetime of shallow-trapped electrons is a major distinction from UiO-66-NH<sub>2</sub> and is the  
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37 result of N atoms within the benzene ring of the pd and pz ligands. If these shallow-  
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39 trapped electrons are accessible for photocatalysis, the longer lifetime becomes a  
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41 significant benefit. Nevertheless, there is a wide temporal gap between these primary  
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43 events (fs–ps) and the oxidation reactions ( $\mu\text{s}$ –s) downstream, which may require  
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45 further investigations to track the electronic dynamics after the system moves out of the  
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47 trap states beyond the ligands, and under more realistic catalytic conditions.

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53 To delineate the contribution of ligands to the overall MOF signal, TA was performed  
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55 on the bare ligands (Figure S6b,c). The high similarity between the ligand and MOF  
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57 TA profiles indicates that the conduction band has considerable ligand character, in  
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3 agreement with PDOS calculations (Figure S4b). The excited state lifetime of the pd  
4 ligand is considerably shorter than that of the corresponding MOF, suggesting that  
5 interactions with the metal nodes and close packing of organic linkers lead to a  
6 prolonged excited state population (e.g., a long-lived trap state).  
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13 Previous literature reported that pristine UiO-66-NH<sub>2</sub> has a very short excited-state  
14 lifetime (~1.5 ps), while samples with missing linker defects show a significantly longer  
15 excited state lifetime (~150 ps).<sup>46</sup> Enhanced ligand-to-metal charge transfer (LMCT)  
16 character can arise from structural defects (e.g., missing linkers), which shrink the  
17 energy gap between the ligand LUMO and metal *d*-orbitals in the MOF. Past a critical  
18 concentration of defects, the excited state lifetime increases dramatically to the ns  
19 timescale, which is thought to correspond to deeply trapped electrons that cannot be  
20 utilized for photocatalysis. Overall, our TA data imply that all four MOFs contain  
21 considerable defects that prolong the excited state lifetimes to hundreds of ps and  
22 facilitate charge transfer, which can benefit HER.  
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37 **Hydrogen Evolution Half-Reaction.** After examining each MOF's photophysical  
38 properties, we turned next to their photocatalytic performance for H<sub>2</sub> evolution. The H<sub>2</sub>  
39 generation experiments were performed using a 305 nm cut-off filter, triethylamine  
40 (TEA) as the sacrificial electron donor, and Pt NPs as the co-catalyst (Figure S7, Table  
41 S1). Upon addition of the Pt NPs with UiO-66 or UiO-66-NH<sub>2</sub>, our TA results show  
42 that there is a lifetime decrease with  $\tau_4$  showing the largest change from 900 to 560 ps  
43 in UiO-66 (Figure S6a) and from 690 to 370 ps in UiO-66-NH<sub>2</sub> (Figure S6b). In  
44 contrast, the TA signals for UiO-66-pd and UiO-66-pz are diminished with Pt NPs due  
45 to destabilization of the MOF suspension. This finding of a shortened excited state  
46 lifetime is in accord with the observed photoluminescence (PL) quenching after mixing  
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Pt NPs with the MOFs (Figure S8). The TA and PL results confirm that Pt NPs attract the electron from the MOFs, decreasing the rate of undesired charge recombination.<sup>24</sup> The photocatalytic reaction conditions were optimized by varying the concentrations of MOF photocatalysts, TEA, and Pt NPs. After optimization, the Pt/UiO-66-NH<sub>2</sub> photocatalytic system showed the highest H<sub>2</sub> evolution rate of ~300 μmol h<sup>-1</sup> g<sup>-1</sup>, followed by Pt/UiO-66-pz and Pt/UiO-66-pd achieving 153 and 97 μmol h<sup>-1</sup> g<sup>-1</sup>, respectively. Pt/UiO-66 was inferior to the other MOFs, displaying the lowest H<sub>2</sub> evolution rate of 34 μmol h<sup>-1</sup> g<sup>-1</sup> (Figure S9). UiO-66-NH<sub>2</sub> and UiO-66-pz possess smaller bandgaps than UiO-66 and UiO-66-pd, permitting more effective light-harvesting and leading to higher H<sub>2</sub> evolution rates. The stability of each MOF post-irradiation was confirmed using PXRD, which showed the preserved crystallinity of each MOF, and the material could be reused without losing catalytic efficiency (Figure S10).

**Dual-Functional Photocatalytic Activity.** To take advantage of both the photoexcited electrons and holes, we replaced the sacrificial electron donor TEA with benzenethiol, envisioning simultaneous production of H<sub>2</sub> and synthesis of sulfide-based products. The oxidation of thiols has been widely used to synthesize sulfide-based products<sup>47-48</sup>, which are essential compounds in biological (drugs synthesis, DNA cleavage)<sup>49</sup> and industrial applications (rubber vulcanization and rechargeable lithium batteries).<sup>50</sup> Therefore, it would be advantageous to develop a dual-functional photocatalytic system that could generate H<sub>2</sub> fuel and concurrently synthesize value-added sulfide-based products (Figure 5a). In the dual-functional reaction, Pt/UiO-66-pz exhibits the highest H<sub>2</sub> evolution rate of 329±13 μmol h<sup>-1</sup> g<sup>-1</sup>, followed by Pt/UiO-66-NH<sub>2</sub> and Pt/UiO-66-pd with rates of 146±2 and 102±16 μmol h<sup>-1</sup> g<sup>-1</sup>, respectively. Pt/UiO-66 shows the

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3 lowest H<sub>2</sub> evolution rate of 55±4 μmol h<sup>-1</sup> g<sup>-1</sup> (Figure 5a). After the photocatalytic  
4 reaction, all the MOFs maintained their crystallinity (Figure S11).  
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8 To determine the efficiency of the oxidation half-reaction, the conversion yield of  
9 benzenethiol to sulfide-based products was quantified by UV-vis spectroscopy using  
10 Ellman's Reagent (dithionitrobenzoic acid, DTNB)<sup>51</sup> (Figure S12). The DTNB peak  
11 intensity at ~410 nm shows the largest decrease for Pt/UiO-66 (Figure 5b), signifying  
12 the largest conversion of benzenethiol to sulfide-based products (75.9%), followed by  
13 Pt/UiO-66-pz and Pt/UiO-66-NH<sub>2</sub> with yields of 70.4% and 53.5%, respectively.  
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15 Pt/UiO-66-pd exhibits the lowest yield at 36.5% compared to the other MOFs. The  
16 large conversion displayed by UiO-66 is likely from the spatial separation of the holes  
17 and electrons in the UiO-66 structure. According to the isodensity surfaces (Figure 3),  
18 the holes are confined to the metal nodes while the electrons are located on organic  
19 ligands. This spatial separation could decrease recombination rates and afford greater  
20 hole mobility, benefitting the oxidation of benzenethiol half-reaction at the expense of  
21 H<sub>2</sub> production. In addition to UV-vis spectroscopy, gas chromatography-mass  
22 spectrometry (GC-MS) was used to qualitatively and quantitatively evaluate the  
23 sulfide-based products with Pt/UiO-66-pz. GC-MS data showed that although our dual-  
24 functional photocatalyst Pt/UiO-66-pz is not selective in the production of diphenyl  
25 disulfide, it can readily oxidize benzenethiol to produce sulfide-based products while  
26 simultaneously producing H<sub>2</sub> (Figure S13).  
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50 The same photocatalytic reaction setup with only Pt NPs and no MOF photocatalyst  
51 showed a lower H<sub>2</sub> evolution rate of 0.75 μmol h<sup>-1</sup> (4.94 μmol h<sup>-1</sup> for Pt/UiO-66-pz)  
52 and sulfide-based products yield of 25.7%, demonstrating that the MOFs are vital for  
53 the photocatalytic system to achieve high H<sub>2</sub> evolution rates and benzenethiol  
54 conversion. Moreover, when the photocatalytic reaction was carried out with D<sub>2</sub>O  
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3 instead of DI-water, Pt/UiO-66-pz had an H<sub>2</sub> evolution rate that was three times lower,  
4 signifying that the H<sub>2</sub> being produced arises from both water (234 μmol h<sup>-1</sup> g<sup>-1</sup>, ~71%)  
5 and dehydrocoupling of the thiols (95 μmol h<sup>-1</sup> g<sup>-1</sup>, ~29%) (Figure S14).<sup>48, 52</sup>  
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11 The photocatalytic reaction was also tested using different wavelengths of light and  
12 different temperatures under different experimental conditions. When a 360 nm light  
13 cut-off filter was used, Pt/UiO-66-pz showed a poor H<sub>2</sub> evolution rate and sulfide-based  
14 products yield (<0.5 μmol h<sup>-1</sup> g<sup>-1</sup>, ~23.3%) (Figure S15). Above 360 nm, UiO-66-pz  
15 showed minimal light absorption, and could not efficiently utilize it for photocatalysis.  
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17 To test for temperature affects, the same reaction setup was used but instead of  
18 irradiating the material, it was place in a 60°C water bath for the same reaction time.  
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20 Under these conditions, Pt/UiO-66-pz had a H<sub>2</sub> evolution rate of <0.5 μmol h<sup>-1</sup> g<sup>-1</sup> and  
21 ~22.2% sulfide-based products yield (Figure S16), demonstrating that the contribution  
22 from temperature was negligible in driving the photocatalytic reaction. Overall, Pt/UiO-  
23 66-pz performs best in the coupled redox system, achieving the highest H<sub>2</sub> production  
24 rate of ~329 μmol h<sup>-1</sup> g<sup>-1</sup> while oxidizing 70.4% of benzenethiol (Figure 5).  
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40 Holes and reactive oxygen species (ROS) are the primary active species responsible for  
41 the photocatalytic oxidation of organic molecules.<sup>53</sup> To elucidate the active species  
42 governing the oxidation of benzenethiol, the hole scavenger triethanolamine (TEOA)  
43 and the hydroxyl radical (•OH) scavenger tert-butanol (tBuOH) were used individually  
44 with Pt/UiO-66-pz in the same photocatalytic setup.<sup>24</sup> Upon adding TEOA, Pt/UiO-66-  
45 pz exhibits an H<sub>2</sub> evolution rate of 233 μmol h<sup>-1</sup> g<sup>-1</sup> and a 62.7% yield of sulfide-based  
46 products, slightly lower than the DFP reaction without a scavenger (Figure 5a, b). Upon  
47 adding tBuOH, the hydrogen evolution rate (173 μmol h<sup>-1</sup> g<sup>-1</sup>) and sulfide-based  
48 products yield (28.6%) decreased significantly. These results confirm that both the  
49 holes and hydroxyl radicals are involved in the oxidation of benzenethiol to sulfide-  
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3 based products, with hydroxyl radicals being the dominant reactive species (Figure  
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6 S17).

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8 To gain further insights into the effect of benzenethiol substituents on both the reduction  
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10 and oxidation half-reactions, we replaced benzenethiol with 4-chlorobenzenethiol,  
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12 which contains an electron-withdrawing chlorine (Cl) atom. In the same DFP reaction  
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14 setup using Pt/UiO-66-pz, we observed a decrease in the H<sub>2</sub> evolution rate (156 μmol  
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16 h<sup>-1</sup> g<sup>-1</sup>) and sulfide-based products yield (36.5%) compared to the DFP reaction with  
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18 benzenethiol (Figure S18). The HOMO of 4-chlorobenzenethiol is slightly lower than  
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20 that of benzenethiol, representing a smaller thermodynamic driving force for hole  
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22 transfer (Figure S19). Moreover, the lower conversion of 4-chlorobenzenethiol could  
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24 stem from the deactivating nature of halogens in electrophilic aromatic substitution  
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26 reactions.<sup>54-55</sup> Assuming that the reaction proceeds via a thiyl radical<sup>48, 56</sup>, the Cl-  
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28 substituent can deactivate the aromatic ring of one radical, making it less susceptible to  
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30 electrophilic attack by another thiyl radical<sup>57</sup>, and thereby decreasing the overall  
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32 reactivity and yield. Despite the reduced yield, Pt/UiO-66-pz is selective in producing  
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34 bis(dichlorophenyl)disulfide, and the dual-functional reaction is still successful. This  
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36 demonstrates that the dual-functional nature of Pt/UiO-66-pz is not limited to a single  
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38 type of benzenethiol reactant.  
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## 45 46 **Conclusions**

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49 Dual-functional photocatalysis enables the utilization of both the photogenerated  
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51 electrons and holes, elevating the photocatalytic system beyond the competition for  
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53 high H<sub>2</sub> production rates and enhancing its utility without the need for sacrificial  
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55 reagents such as TEA. This work employed four MOFs (UiO-66 analogs) with N-  
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57 functionalized ligands in a dual-functional photocatalytic reaction. The number and  
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3 position of the N-atoms attached to or within the benzene ring correlate to the  
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5 photophysical (light absorption, bandgap, and excited state dynamics) and electronic  
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7 (conductivity) properties of each MOF. We demonstrated that the presence of the N-  
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9 atoms in the ligands effectively enhances the photophysical properties of MOFs toward  
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11 efficient DFP.  
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15 The Pt/Uio-66-pz manifests the highest H<sub>2</sub> evolution rate and the second-best  
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17 benzenethiol conversion. Compared to other MOFs, the higher performance of Uio-  
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19 66-pz can be attributed to an ultrafast energy stabilization step (~4 ps) toward a shallow  
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21 trap state (~95 ps lifetime) to facilitate charge transfer, a relatively small bandgap (3.1  
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23 eV), and high conductivity. Although the structure and functionality of Uio-66-pd was  
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25 closely similar to Uio-66-pz, the lower absorption range and electron conductivity of  
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27 Uio-66-pd leads to the poorer performance on both H<sub>2</sub> evolution rate and the  
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29 benzenethiol conversion. In contrast, Uio-66 performs the best in benzenethiol  
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31 conversion (75.9%) due to the spatial separation of electrons and holes but is the worst  
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33 photocatalyst for H<sub>2</sub> generation due to its wider bandgap, lower conductivity, and short  
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35 excited-state lifetime. Thus, MOFs for DFP need a moderately small bandgap that  
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37 straddles the water oxidation and reduction potentials, a VB that is below the HOMO  
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39 of the organic molecule, high conductivity, and shallow trap states that allow for longer  
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41 (~100 ps) excited state lifetimes. Our work has improved our understanding of the  
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43 interplay of each of these features, allowing for the design and development of more  
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45 efficient dual-functional photocatalytic MOFs through careful consideration of each of  
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47 these parts.  
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55 Our study characterizes the photophysical properties that underlie highly efficient DFP,  
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57 especially for Pt/Uio-66-pz, wherein both the production of H<sub>2</sub> and synthesis of organic  
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59 molecules can be achieved simultaneously. With the deepened mechanistic insights  
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3 from ultrafast exciton dissociation, charge transfer, and predictive knowledge about  
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5 frontier orbitals containing ligand  $\pi^*$  and metal *d*-orbitals, we envision future rational  
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7 design of MOFs to optimize the yield and selectivity of dual-functional photocatalytic  
8  
9 reactions. A potential design route is to further red-shift the absorption of N  
10  
11 heterocycles using electron-donating substituents (e.g.,  $-\text{NH}_2$ ,  $-\text{OH}$ ) or extending the  
12  
13 ligand structure with dipyrazine ligands. The enhanced excited-state relaxation pathway  
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15 out of a shallow-trap state could be coupled with a broader absorption profile to  
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17 maximize the DFP performance, augmented by fine-tuning the HOCO-LUCO  
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19 alignment for these new N-heterocyclic MOFs.  
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### 29 **Supporting Information**

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33 The Supporting Information is available free of charge at <https://xxx>. Experimental  
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35 methods, femtosecond transient absorption, computational methods, photocatalysis  
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37 experiments and analysis are provided in the Supporting Information.  
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### 42 **Author Contributions**

43  
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45 K.C.S. conceived the project and designed the experiments together with N.C.C., and  
46  
47 M.T.N. N.C.C. led the experimental work. N.C.C. and M.T.N. performed the DFP  
48  
49 experiments and interpreted the data. S.K.W. synthesized the materials used for DFP.  
50  
51 L.T., L.S.L., and C.F. performed the transient absorption experiments and analyzed the  
52  
53 data. J.S.H. and T.Z. led the theoretical calculations described in this work. E.M.H. and  
54  
55 K.G. analyzed the oxidation products formed after DFP. All authors contributed to the  
56  
57 writing and editing of the manuscript.  
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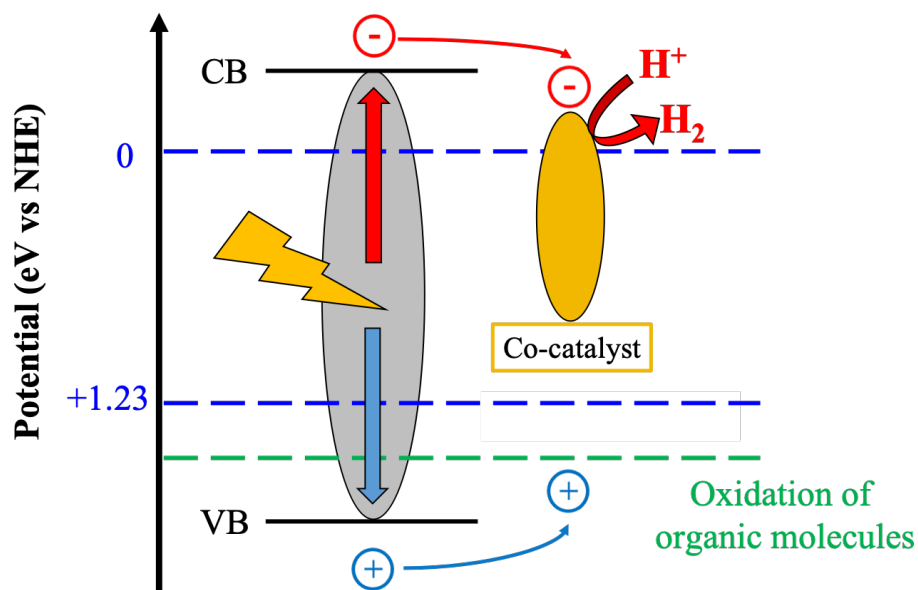
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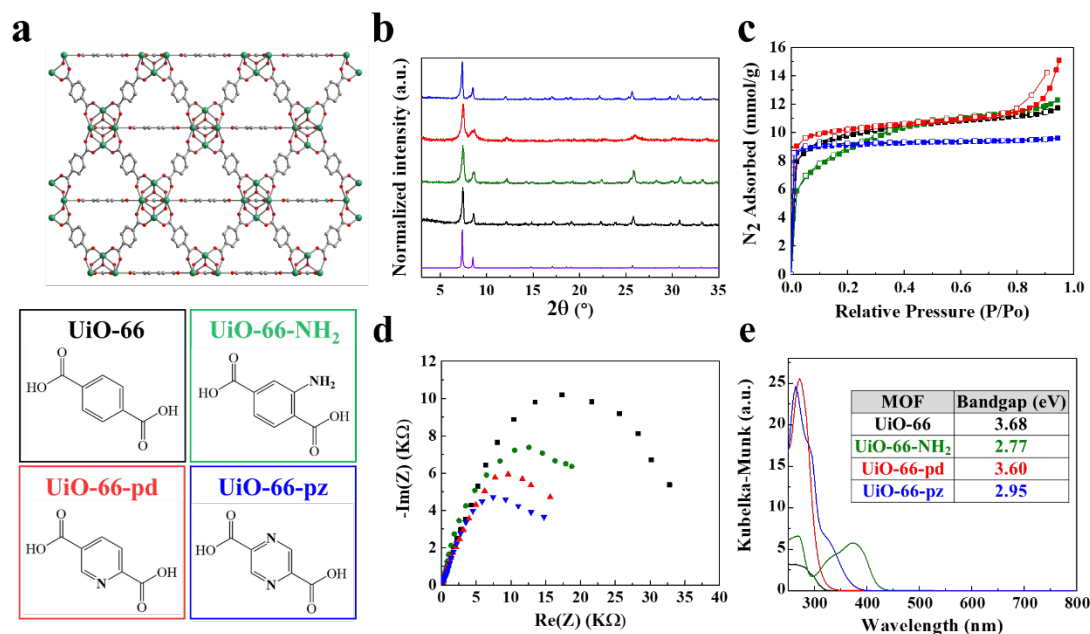
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## Figures

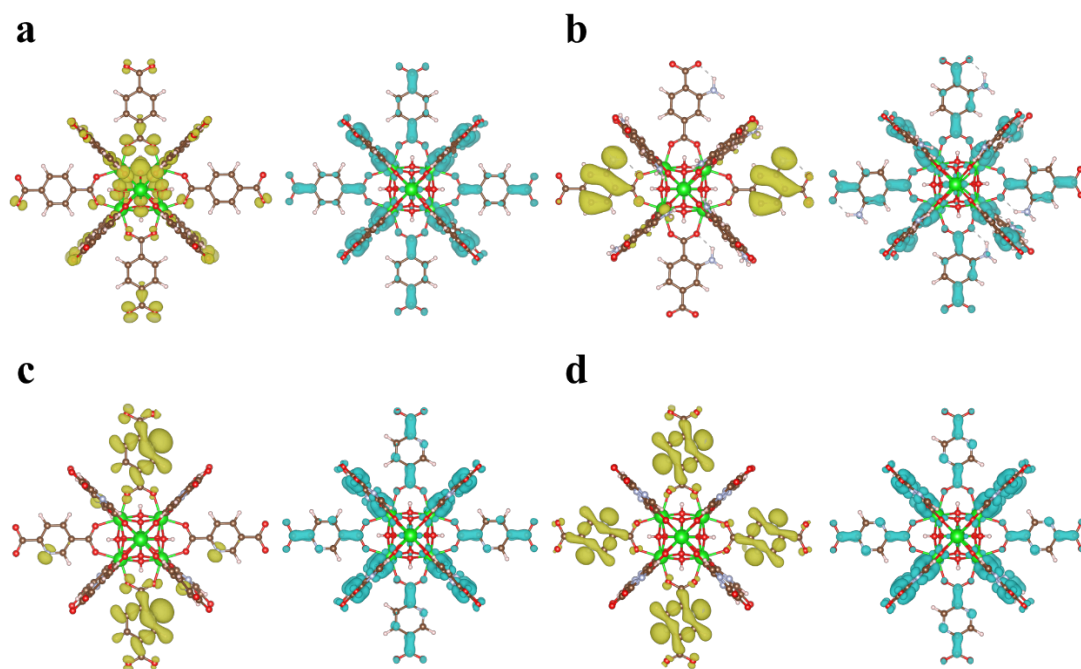


**Figure 1.** Schematic of a dual-functional photoredox system. Upon irradiation of the MOF, an electron is excited to the conduction band (CB) while a hole is left in the valence band (VB). The electrons in the CB of the photocatalyst are transferred to the co-catalyst (Pt NPs), which is responsible for reducing  $H^+$  to  $H_2$  ( $H^+/H_2$ , 0.0 eV vs. NHE). Meanwhile, the holes in the VB of the photocatalyst oxidize the organic molecule by moving to its HOMO.

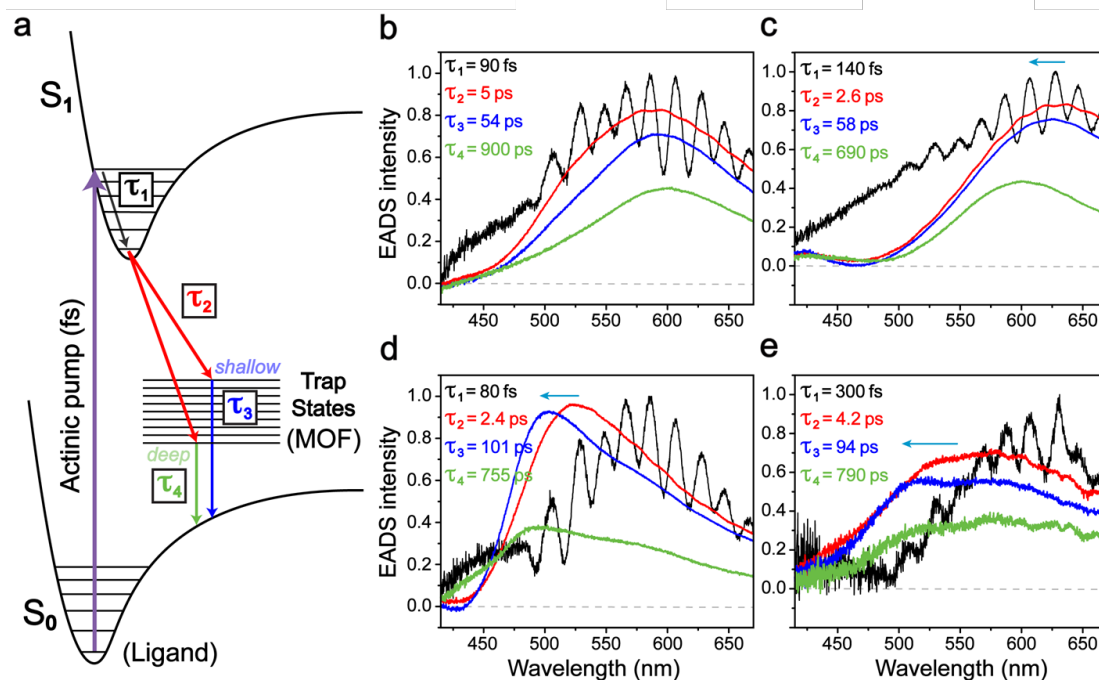


**Figure 2.** Structures and characterization of the four UiO-66 analogs. a) Representative structure of UiO-66 (top panel) and organic ligands used to synthesize four UiO-66 analogs, including terephthalic acid (black), 2-aminoterephthalic acid (green), 2,5-pyridinedicarboxylic acid (red), and 2,5-pyrazinedicarboxylic acid (blue). b) PXRD patterns of all four MOFs match the simulated UiO-66 PXRD pattern (purple), which can be synthesized as phase pure. c) Type-I N<sub>2</sub> isotherms collected for all MOFs at 77 K reveal that they are permanently porous and adsorb comparable amounts of N<sub>2</sub>. d) EIS Nyquist plots reveal that UiO-66-pz has the smallest semi-circle radius, representing the lowest charge-transfer resistance. e) Kubelka–Munk transformation of the MOF’s diffuse reflectance spectra collected from 250 to 800 nm. Inset: calculated bandgaps.

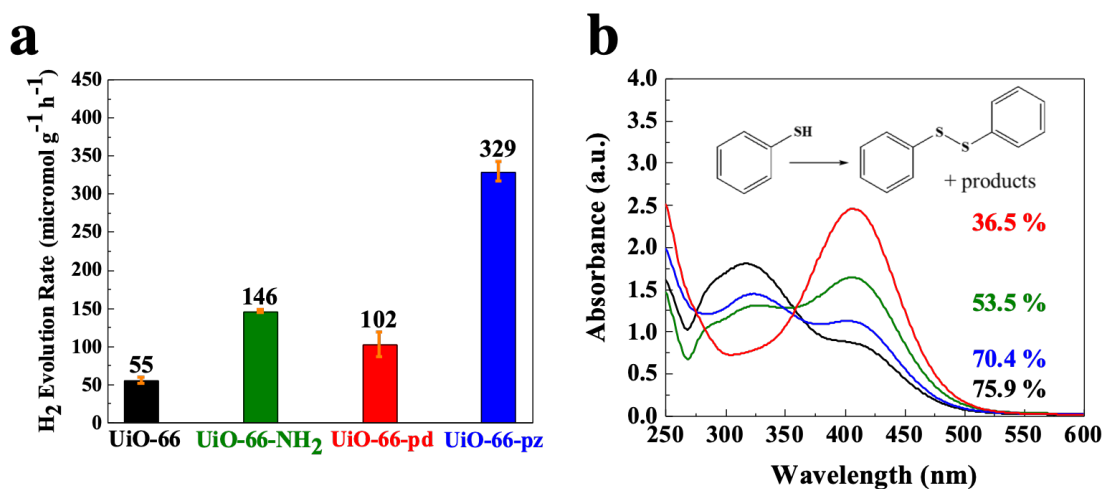




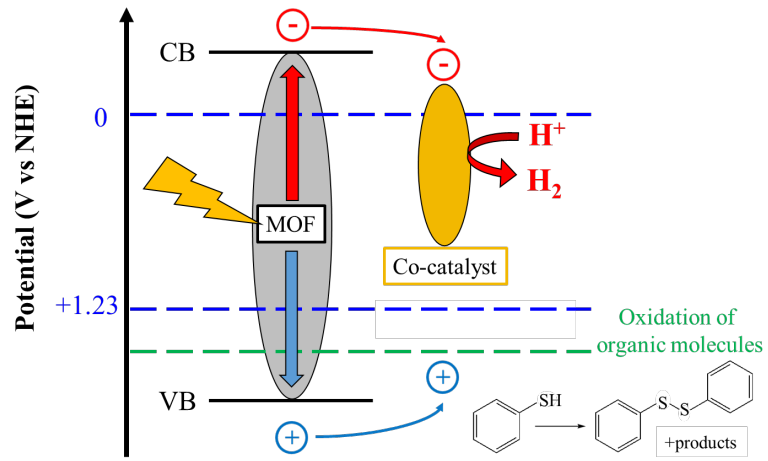
**Figure 3.** Electron-hole isodensity surfaces for the four UiO-66 analogs computed using density-functional theory (DFT). The  $\rho_{h^+}(r)$ -region (yellow) represents hole accumulation, and the  $\rho_{e^-}(r)$ -region (blue) indicates electron accumulation. The  $\rho_{e^-}(r)$ -region for a) UiO-66, b) UiO-66-NH<sub>2</sub>, c) UiO-66-pd, and d) UiO-66-pz, is comparable and located mainly on the ligand. The  $\rho_{h^+}(r)$ -region of UiO-66 is positioned on the nodal and linker oxygens, whereas the  $\rho_{h^+}(r)$ -region for UiO-66-NH<sub>2</sub>, UiO-66-pd, and UiO-66-pz is positioned mostly on the ligand.



**Figure 4.** Transient electronic dynamics of the four UiO-66 analogs in suspension. a) Schematic of potential energy surfaces of the organic ligand and intermediate (both shallow and deep) trap states in MOFs. Key excited state energy relaxation pathways as primary events are depicted with the associated time constants ( $\tau_1$  to  $\tau_4$ ). Global analysis with a sequential kinetic model of the fs-TA spectra of b) UiO-66, c) UiO-66-NH<sub>2</sub>, d) UiO-66-pd, and e) UiO-66-pz yields four characteristic lifetimes (listed in the insets):  $\tau_1$ , black;  $\tau_2$ , red;  $\tau_3$ , blue; and  $\tau_4$ , green for the evolution-associated difference spectra (EADS, normalized at the highest peak intensity for the initial black trace), which track the excited-state absorption (ESA) band dynamics. Except for UiO-66, a notable ESA peak blueshift on the few-ps timescale is depicted by cyan arrows in (c-e).



**Figure 5.** Dual-functional photocatalytic hydrogen generation and oxidation of benzenethiol. a) Comparison of H<sub>2</sub> evolution rates for UiO-66 (black), UiO-66-NH<sub>2</sub> (green), UiO-66-pd (red), and UiO-66-pz (blue) mixed with Pt NPs and 0.1 mL benzenethiol. b) UV-vis absorption spectra show the decrease of the DTNB peak at ~410 nm tracking the conversion to sulfide-based products with the best performance by Pt/UiO-66 (black), followed by Pt/UiO-66-pz (blue), Pt/UiO-66-NH<sub>2</sub> (green), and Pt/UiO-66-pd (red). The conversion yields were calculated using the calibration curve shown in Figure S20.



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