

Photooxidation of Water with Thin Film Tungsten Oxides

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Cite This: *J. Chem. Educ.* 2021, 98, 614–619

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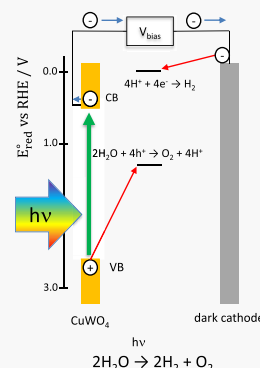
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ABSTRACT: Tungsten oxide semiconductors are used to photooxidize water. Students prepare WO_3 , CuWO_4 , and mixed composition thin film photoanodes. The syntheses are simple, allowing for the quick production of a range of materials. Photoanodes are tested in a photoelectrochemical cell assembled from common equipment, allowing for the direct comparison of photoanode performance. Composite photoanodes perform best, leading to a discussion of what limits overall performance in these materials. Additional instrumental methods such as SEM, EDS, XRD, and diffuse reflectance UV–vis spectrophotometry, if available, can be incorporated readily into the experiment, providing additional insights into material properties. Students are introduced to solid-state chemistry, semiconductors, and the photooxidation of water in the context of the hydrogen economy.



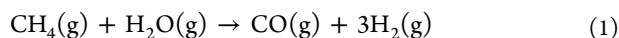
KEYWORDS: Upper-Division Undergraduate, Inorganic Chemistry, Interdisciplinary/Multidisciplinary, Laboratory Instruction, Hands-On Learning/Manipulatives, Electrochemistry, Instrumental Methods, Materials Science, Photochemistry, Semiconductors

INTRODUCTION

There is enormous potential for using the energy in sunlight to split water, producing molecular hydrogen fuel.¹ While molecular hydrogen is the high-value product in this scheme, the coevolution of oxygen gas is the more difficult reaction to drive. Photoelectrochemical cells (PECs) use semiconductor electrodes that harvest sunlight and facilitate the splitting of water.² The material properties of the electrodes employed in PECs are critical to their function.³

Proponents of the hydrogen economy envision hydrogen gas as a low-carbon fuel for heating and transportation, and as a sink for energy storage and long-distance energy transportation. The high gravimetric energy density stored in the hydrogen molecule can be converted directly into electrical energy with high efficiency via a fuel cell. Hydrogen gas is clean burning, producing water as its sole combustion product. Nonetheless, the carbon footprint of the hydrogen economy will be determined by the ultimate source of hydrogen.

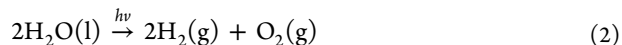
Molecular hydrogen does not occur naturally in large reservoirs on Earth. Its manufacture requires a hydrogen source, or carrier, such as a hydrocarbon fossil fuel or water. The majority of hydrogen produced today comes from steam methane re-forming, a process that consumes a hydrocarbon hydrogen carrier and creates carbon monoxide:



However, the preponderance of hydrogen atoms in the biosphere is carried in water. A process to split water,

liberating molecular hydrogen and molecular oxygen and driven by renewable energy, would provide a truly green fuel.

In this context, photoelectrochemistry has been considered a promising approach to split water into hydrogen and oxygen gases.⁴ The photoelectrochemical splitting of water can be performed by combining a photoanode with a dark cathode. In general, metal oxide semiconductors are used as photoanodes to evolve oxygen, and metals or photocathodes are used to evolve hydrogen gas. Overall, sunlight powers the carbon-free production of hydrogen gas:



Fujishima and Honda⁵ were photoelectrochemical pioneers in the early 1970s, using wide band gap, n-type TiO_2 to split water, while Hodes et al.⁶ introduced WO_3 a few years later. This *Journal* documented the early advances in PECs for the direct conversion of sunlight into chemical fuels such as hydrogen,^{7–9} while recent reports address the electrochemistry of semiconductor-assisted photooxidation^{10,11} and the catalysis of water photooxidation.^{12–16} All of this work is grounded in a firm understanding of electrocatalysis in water splitting,¹⁷ the

Received: July 20, 2020

Revised: November 30, 2020

Published: December 23, 2020



insights provided by electrochemical techniques,¹⁸ the energetics of photochemical reactions,¹⁹ and an understanding of photo quantum yield.²⁰

Semiconductors suitable as electrodes for the photoelectrolysis of water with sunlight require the following:²¹

- moderate conductivity (preferably due to a combination of large carrier mobility and small carrier concentration);
- a minimal number of recombination centers;
- chemical stability both in the dark and under illumination;
- a band gap of approximately 1.8–2.1 eV to maximize solar radiation absorption and drive the reaction at a rate sufficient for practical fuel production; and
- to a first approximation, suitable positioning of the conduction and valence band edges relative to the H^+/H_2 and $\text{O}_2/\text{H}_2\text{O}$ reduction potentials, respectively.

While these requirements have been understood for almost 50 years, there is no ideal candidate meeting all of these criteria. Of known compounds, metal oxides show promise in a number of these areas and have been studied extensively. Many such as TiO_2 have the required long-term stability for practical use.²² Some such as $\alpha\text{-Fe}_2\text{O}_3$ have band gaps that extend into the visible.²³ WO_3 and CuWO_4 are metal oxide semiconductors of moderate conductivity with few recombination centers and good-to-excellent chemical stability.²⁴ The fifth condition speaks to the spontaneity of the water splitting reaction for a PEC wherein the semiconductor electrode is equilibrated directly with the hydrogen and oxygen evolution reactions. This condition is not absolute, and in cases where the band alignment is not ideal, it is still possible to split water with the addition of a small amount of electrical energy to supplement the energy derived from sunlight. A few metal oxide semiconductors remain good candidates for use as photoanodes in light-assisted water splitting in this limit.

The band diagrams for WO_3 and CuWO_4 , wide band gap n-type semiconductors, are shown in Figure 1. The band gap,

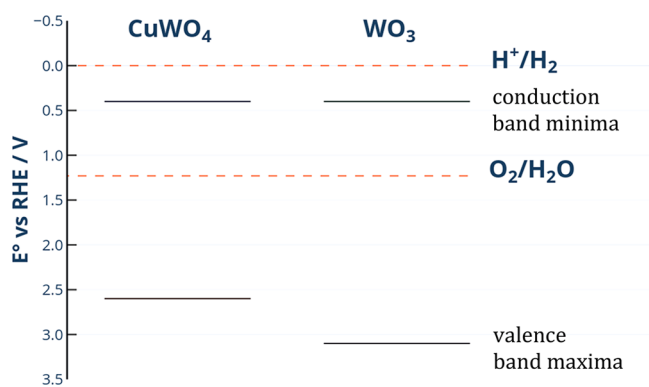


Figure 1. Band diagrams for WO_3 and CuWO_4 showing the standard reduction potentials for H^+ and O_2 . The positions of the WO_3 and CuWO_4 band edges are taken from the literature.^{27,29}

equal to the difference between the conduction and valence band edges, is typically determined from optical measurements, and the position of the conduction band edge is determined from electrochemical impedance spectroscopy.²⁵ Their conduction bands, largely of $\text{W}(5\text{d})$ character, are at the same potential,^{26,27} slightly more positive than the standard reduction potential of protons. The valence bands, with $\text{O}(2\text{p})$ and $\text{Cu}(3\text{d}) + \text{O}(2\text{p})$ character, respectively, have edges

significantly positive of the standard reduction potential of oxygen, $E^\circ = +1.23 \text{ V}$ vs NHE at $\text{pH} = 0$. The 2.7 eV band gap of WO_3 is on the edge of the visible, whereas CuWO_4 with a smaller band gap of 2.2 eV will absorb more solar radiation. WO_3 is stable in solutions having a pH below 7, whereas the chemical stability of CuWO_4 at moderate to high pH is superior.^{27,28}

Figure 2 shows CuWO_4 in a PEC photooxidizing water with the addition of a low applied potential to enable the reduction

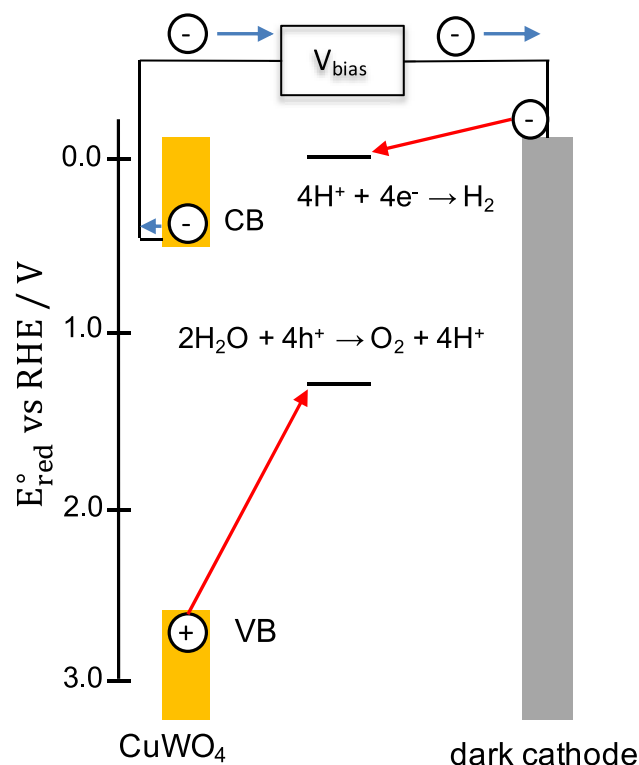
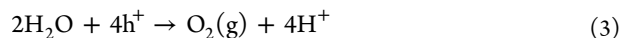


Figure 2. Energy diagram for a PEC with a CuWO_4 photoanode and a dark cathode. A photon with energy greater than the band gap of CuWO_4 has been absorbed by the photoanode, producing the electron/hole pair at the edges of the conduction (CB) and valence bands (VB). The transport of electrons is shown with blue arrows while interfacial electrochemistry due to electron or hole transfer is shown in red. The hole oxidizes water at the photoanode while the electron is transferred via the circuit to the dark cathode, where, with the addition of a small amount of electrical energy, the electron reduces protons.

of protons at the dark cathode. The CuWO_4 photoanode absorbs light, promoting an electron from the valence band to the conduction band and leaving a hole in the valence band. Given the position of the valence band edge relative to the reduction potential of oxygen, holes (h^+) in the valence band are sufficiently energetic such that they can react with water to produce oxygen gas:



However, an electron at the conduction band edge is not sufficiently energetic to react with protons in water to produce hydrogen gas:



Nevertheless, the electron can be transferred via the external contact of the photoanode to the dark cathode, and the

potential of the electron can made more negative than the reduction potential for protons with a small input of electrical energy, allowing reaction 4 to proceed. In terms of the overall energetics, the majority of the energy for splitting water is supplied by the photon.

Practical metal oxide photoanodes are fabricated as thin films. Keeping the film thickness smaller than the hole diffusion length allows photogenerated carriers throughout the semiconductor to move to the surface where they are harvested without excessive loss. Utilizing thin films also minimizes the total amount of material required when constructing large area devices. While WO_3 was typically formed through the anodization of tungsten metal,⁶ thin films of both WO_3 and CuWO_4 can be produced easily via electrodeposition,^{27,28,30} spin coating,³¹ or drop-casting³² from W(VI) precursors prior to calcination in air at moderate temperatures.

While a deep exploration of the five criteria for effective photoelectrolysis listed above is not possible in the course of a single experiment, the interplay of absorbance, electron–hole recombination, and electrocatalysis across WO_3 , CuWO_4 , and a range of their composites can be examined in PEC test cells, yielding insights into the material properties required for the effective splitting of water with sunlight.

■ EXPERIMENTAL OVERVIEW

The production of photoanodes with varying compositions, followed by characterization in a PEC, is at the core of this experiment. Detailed procedures for the production and characterization of photoanodes are provided in the [Supporting Information](#) along with a list of required equipment and chemicals. The only required, nonroutine equipment is a programmable furnace capable of reaching 550 °C and a potentiostat.

Thin films of WO_3 , CuWO_4 , and mixed composition $\text{CuWO}_4/\text{WO}_3$ semiconductors are prepared in parallel by drop-casting solutions of ammonium metatungstate hydrate and copper nitrate trihydrate salts in ethylene glycol solution onto fluorine-doped tin oxide (FTO) substrates.³² The substrates are dried at 120 °C in air for 30 min and then transferred to a programmable furnace for calcination at 550 °C in air.

A PEC test rig is assembled from

- a potentiostat;
- a reference electrode;
- a carbon rod;
- a suitable optical source;
- a 150 mL beaker;
- a lab jack; and
- a ring stand with associated clamps.

Linear scan voltammetry (LSV) is used to collect current–voltage curves under simulated solar irradiation and in the dark. While the data presented here were acquired with an AM1.5G source that reproduces solar irradiance, a ceramic metal halide grow light is a highly effective, readily available, low-cost substitute. Alternately, either a bright metal halide desk lamp or a slide projector is a source sufficient to accomplish the goals of the experiment.

Extensions

The photoanodes are readily amenable to characterization by a number of instrumental techniques. Energy dispersive X-ray spectroscopy (EDS) probes elemental composition and is particularly useful when working with composite samples. X-

ray diffraction (XRD) provides information on crystal structure, unit cell dimensions, and crystallite size. Secondary electron scanning electron microscopy (SEM) details the surface morphology including the dimensions of mesopores. Diffuse reflectance UV–vis spectrophotometry addresses the interaction of the nanostructured thin films with visible light.

In addition, an electrocatalyst for the oxygen evolution reaction (OER) greatly enhances performance. A manganese phosphate electrocatalyst incorporated onto the surface of the photoanode can increase the overall rate of oxygen evolution by up to 50%.³²

Logistics

The experiment, typically run with students working in pairs, has been offered five times in an upper-division instrumental methods laboratory, in an upper-division materials chemistry hybrid laboratory/tutorial, and in a first-year introduction to chemical research laboratory. One 4 h laboratory session is dedicated to production of the photoanodes, with 2 h on a subsequent day set aside for PEC characterization. An additional 2 h session allows for roughly six samples to be characterized by SEM and EDS, while four samples can be characterized by high-resolution XRD at the same time.

■ HAZARDS

Cutting glass often produces small glass shards. Wear gloves when handling sheets of substrates and cutting glass. Sweep the working area benchtop immediately after cutting glass. AM1.5G and grow light sources emit ultraviolet light. Use appropriate shielding, and minimize exposure of eyes and skin.

■ RESULTS AND DISCUSSION

Figure 3 shows current–voltage curves for the two homogeneous semiconductor photoanodes and a composite photoanode of 1:4 $\text{CuWO}_4:\text{WO}_3$. Positive current represents oxidation at the photoanode. Light from the optical source is alternately allowed to illuminate the cell or is blocked via insertion of a cardboard shutter by hand at roughly 0.2 Hz to provide an internal, dark current baseline for each run. Negligible current flows in the dark for all three materials, while under illumination considerable current is seen starting at potentials well negative of E_{O_2} , which is +0.738 V vs Ag/AgCl/sat. KCl in the 0.1 M Na_2SO_4 electrolyte at pH 5. The photocurrent to oxygen conversion efficiency at WO_3 in this electrolyte exceeds 88%;²⁸ thus, photocurrent is a convenient measure of oxygen gas production.

All three materials photooxidize water at E_{O_2} . The application of increasingly positive potential produces even more oxygen as the activation barrier for the OER is surmounted in this system that is not mass transport limited. Most importantly, oxygen produced under illumination at potentials well negative of E_{O_2} is clear evidence of solar light-to-chemical energy conversion for all three materials; the energy of photons is harvested to drive the OER at applied potentials where it would not occur at a dark electrode.

Oxygen production for a photoanode is a function of absorbance, carrier separation and transport, and electrochemical kinetics at the semiconductor/solution interface. The energy of light absorbed by a material is governed by its band gap, which describes how far into the spectrum of visible light the material will harvest photons. The amount of light absorbed is governed by the material's absorption coefficient,

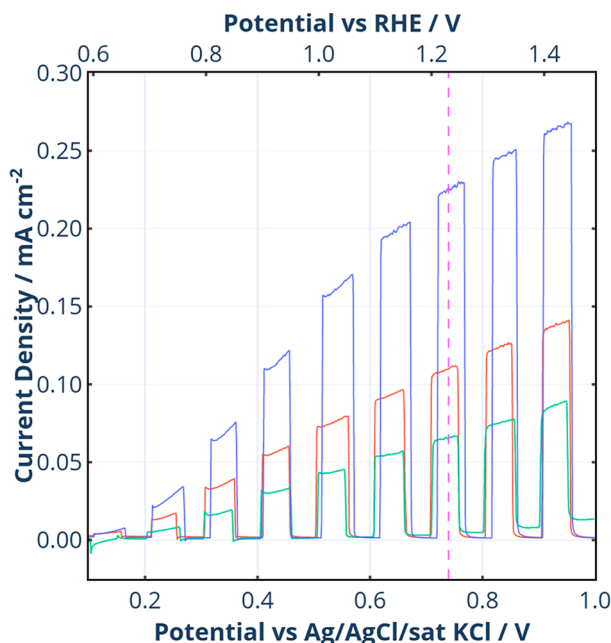


Figure 3. Linear scan voltammograms for photoanodes composed of WO_3 (red), CuWO_4 (green), and 1:4 $\text{CuWO}_4:\text{WO}_3$ (blue) in 0.1 M Na_2SO_4 at pH 5 illuminated with an AM1.5G source at 100 mW cm^{-2} . The scan is initiated at +0.1 V vs Ag/AgCl/sat. KCl at a rate of 25 mV s^{-1} , and the source intensity is chopped by hand at roughly 0.2 Hz. The dashed line marks E_{O_2} .

which describes how many photons are absorbed per unit thickness, and the material's surface morphology, which influences how much light is scattered rather than absorbed. Once a photon has been absorbed and an electron/hole pair created, these charged carriers must be separated, with the hole moving to the semiconductor/solution interface and the electron to the back of the electrode where it is shuttled into the external circuit and transferred to the dark cathode. If the electron/hole pair is allowed to recombine, then the electrochemical water splitting reaction will not be achieved and the energy of the absorbed photon will be converted to heat. Last, once the hole is delivered to the semiconductor/solution interface, the electrochemical kinetics of the OER determine the amount of oxygen produced.

The figure of merit for photoanode performance is the current density at E_{O_2} under illumination. As shown in Table 1,

Table 1. Thin Film Photoanode Average Current Densities at E_{O_2} under AM1.5G Illumination

composition	$j/(\text{mA cm}^{-2})^a$
WO_3	0.105 ± 0.004
CuWO_4	0.058 ± 0.006
1:4 $\text{CuWO}_4:\text{WO}_3$	0.232 ± 0.016

^aStandard deviation values for $n = 3$.

WO_3 performs slightly better than CuWO_4 for the drop-cast electrodes produced here. Considering only absorbance, this is counterintuitive, given that CuWO_4 with its smaller band gap should utilize more of the solar irradiance with its onset of absorbance at 540 nm vs 460 nm for WO_3 . When seeking an absorbance-based explanation, the relative behavior of the pair could be due to differences in crystallite size (instructor notes,

Figure S6), with greater light scattering occurring at the more highly textured CuWO_4 decreasing the number of photons absorbed. With the consideration of carrier separation and transport, superior WO_3 photocurrents could be explained by the drop-cast CuWO_4 having lower carrier mobility and greater recombination, in effect giving up a greater fraction of carriers than WO_3 . In terms of overall performance of pure, thin film photoanodes, the relative photocurrents observed here are consistent with reports of drop-cast WO_3 and CuWO_4 showing roughly identical photooxidation current densities.³²

Composite photoanodes produce considerably more photo-oxidation than their pure constituents. The absorbance and electrochemical kinetics of the individual constituents are unlikely to change when the materials are combined in the composite. Thus, the improved performance of composites is likely due to enhancements in carrier separation and transport therein. Nam et al. propose a two-step mechanism to explain this behavior.³² In the first step, holes formed in the valence band of WO_3 transfer to the valence band of CuWO_4 , while in the second step electrons in the conduction band of CuWO_4 transfer to the conduction band of WO_3 . The driving force for the first step is readily apparent in Figure 1: holes formed in the WO_3 valence band can “bubble upwards” roughly 0.5 eV by moving from WO_3 to CuWO_4 . While there is no net driving force for the second step, it is akin to a sensitization of the larger band gap semiconductor by the smaller, wherein electrons from the CuWO_4 are injected into the WO_3 . The net result is a segregation of carriers in adjacent materials, leading to decreased recombination. An alternate mechanism consistent with the data involves enhanced carrier transport wherein interfacial defect centers at grain boundaries in the composite materials are healed by the second material present, producing longer carrier diffusion lengths and less recombination.

Finally, with regard to electrochemical kinetics, the limiting reaction in the PEC is the OER. When paired with the hydrogen evolution reaction (HER) at the dark cathode, water is electrolyzed with a photoassist. Of the two reactions, the OER is the more difficult to accomplish. This experiment focuses on driving the OER with a photoassist. While the surfaces of the photoanodes produced here have some photocatalytic capacity, the addition of an electrocatalyst, $\text{Mn}_5(\text{PO}_3(\text{OH}))_2(\text{PO}_4)_2(\text{H}_2\text{O})_4$, greatly enhances the OER. (See the extensions in the instructor notes.) In this fashion, the material properties of the photoanode suitable for the photooxidation of water can be addressed separately.

Extensions

Additional instrumental techniques, if available, greatly aid in the characterization of the photoanodes. EDS of WO_3 photoanodes shows signals from only W and O, with Sn from the FTO substrate also present (Figure S7). CuWO_4 shows equal atomic percentages of Cu and W (Figure S8), while the ratio of Cu:W is 1:4 in a mixed composition photoanode produced from a 1:4 ratio of metals in the drop-casting solution (Figure S9).

XRD shows that the WO_3 produced after calcination at 550°C is monoclinic,³³ whereas the CuWO_4 is triclinic²⁹ (Figure S5). Composite photoanodes show both WO_3 and CuWO_4 features. The lines observed can be indexed from calculated crystal structures, either directly via databases such as The Materials Project,^{34,35} or with software such as VESTA³⁶ or CrystalMaker³⁷ to visualize the unit cell from experimental

data in The American Mineralogist Crystal Structure Database,³⁸ and VESTA or CrystalDiffract³⁷ subsequently to calculate powder patterns.

SEM shows that WO₃ thin films form a smooth surface of 20–50 nm crystallites, whereas the CuWO₄ forms an open porous system of 10–20 nm diameter crystallites. Composite films show congealed WO₃ islands in a porous sea of smaller CuWO₄ crystallites (Figure S6).

Diffuse reflectance UV–vis spectrophotometric data correlates nicely with band gaps and the observed colors (Figure S10).

STUDENT LEARNING

This experiment focuses on understanding the effects of band gap, charge separation and transport, and interfacial electron transfer kinetics of the OER on the ability of metal oxide photoanodes to split water using evidence from a PEC and various instrumental techniques. Learning objectives were assessed via a rubric applied to final reports and posters. Students demonstrated a strong understanding of and the ability to explain the effects of band gap and charge separation and transport upon the production of oxygen at a photoanode under illumination using data collected from electrochemical techniques. Their understanding of the role of interfacial electron transfer kinetics for the OER on the production of oxygen was weak in the first offering of this experiment. Subsequent revisions in the introductory materials, as well as the addition of an informal white board talk on this subject during down time on the first day of the experiment, greatly enhanced student learning on this point. Finally, students were able to integrate data from multiple instrumental techniques to support their arguments.

Students self-identified challenges in recalling and applying concepts of cell potentials. Students were almost uniformly appreciative of the ability to synthesize a material that could split water with sunlight, and many found the opportunity to apply multiple techniques to a solid-state system to form a more complete, unifying picture gratifying.

SUMMARY

Thin film photoanodes of WO₃ and CuWO₄ can be easily fabricated in a short time and their performance in the photooxidation of water measured with a simple photoelectrochemical cell. While additional instrumental techniques lend insights into material properties, the core content of this experiment allows students an opportunity to observe how semiconductor properties affect the performance of a photoanode in a key reaction at the heart of the hydrogen economy.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available at <https://pubs.acs.org/doi/10.1021/acs.jchemed.0c00976>.

Experimental protocol including introduction and student directions (PDF, DOCX)

Notes for instructors (PDF, DOCX)

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

The authors thank Adam R. Johnson for helpful discussions. The National Science Foundation supported acquisition of the scanning electron microscope through DMR-1126080.

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