

1 kV Self-Aligned Vertical GaN Superjunction Diode

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Abstract— This work demonstrates vertical GaN superjunction (SJ) diodes fabricated via a novel self-aligned process. The SJ comprises n-GaN pillars wrapped by the charge-balanced p-type nickel oxide (NiO). After the NiO sputtering around GaN pillars, the self-aligned process exposes the top pillar surfaces without the need for additional lithography or a patterned NiO etching which is usually difficult. The GaN SJ diode shows a breakdown voltage (BV) of 1100 V, a specific on-resistance (R_{ON}) of $0.4 \text{ m}\Omega\cdot\text{cm}^2$, and a SJ drift-region resistance (R_{dr}) of $0.13 \text{ m}\Omega\cdot\text{cm}^2$. The device also exhibits good thermal stability with BV retained over 1 kV and R_{ON} dropped to $0.3 \text{ m}\Omega\cdot\text{cm}^2$ at 125°C . The trade-off between BV and R_{dr} is superior to the 1D GaN limit. These results show the promise of vertical GaN SJ power devices. The self-aligned process is applicable for fabricating the heterogeneous SJ based on various wide- and ultra-wide bandgap semiconductors.¹

Index Terms— power electronics, gallium nitride, nickel oxide, superjunction, charge balance, self-align, high voltage

I. INTRODUCTION

Superjunction (SJ) is arguably one of the most innovative device structures for power electronics [1]. It consists of alternative n- and p-type regions with the balanced charge, which engineer electric field (E-field) in an additional direction normal to the current conduction [2]. In a vertical SJ device, the charge balance allows for high doping concentration while retaining a uniform blocking E-field, enabling the specific on-resistance ($R_{ON,SP}$) to linearly increase with breakdown voltage (BV). In contrast, 1D devices are limited by a trade-off that $R_{ON,SP}$ increases with the square of BV [1].

SJ has achieved a success in Si [3] and was recently reported in SiC [4]–[6]. It promises a superior limit in GaN. However, there still lacks experimental demonstration of a full GaN SJ due to fabrication challenges. Despite the recent progress on p-GaN regrowth [7] and ion implantation [8], the accurate charge control in the selective-area p-GaN is still difficult. Meanwhile, SJ concept has been leveraged in lateral GaN devices based on polarization charges [9]–[13]; but these devices are still limited by a $R_{ON,SP} \propto BV^2$ trade-off [2].

An alternative approach to make the SJ is pairing the n-type semiconductor with a distinct p-type material. Nickel oxide (NiO) is a candidate p-type material for GaN, as its bandgap (3.4–4 eV) and projected critical E-field (5 MV/cm [14]) are higher than GaN and it can form non-leaky junctions on GaN

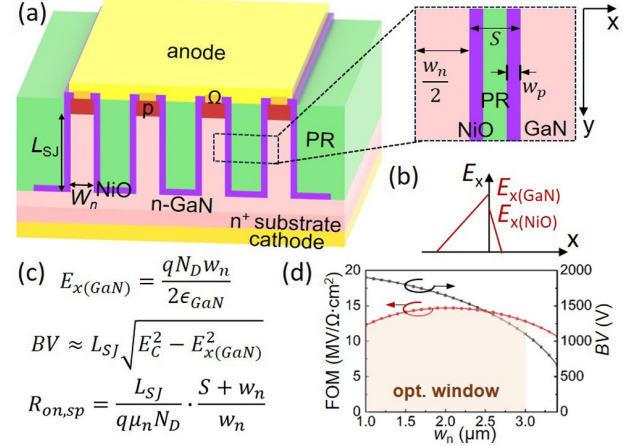


Fig. 1. (a) Schematic of the vertical GaN/NiO SJ diode and a SJ unit-cell. (b) Schematic of the lateral E-field distribution. (c) Major models of the BV and $R_{ON,SP}$ of the SJ region. (d) Modeled SJ FOM and BV as a function of w_n .

[15]. Recently, a tunable acceptor concentration (N_A) is realized in p-NiO, enabling fabrication of a vertical GaN heterojunction-based SJ [16].

However, the SJ fabrication in [16] is very complicated mainly due to the difficulty of NiO etch. It involves two steps of deep GaN etch and multiple alignments to the narrow n-GaN pillars. Besides, the outer sidewalls of periphery GaN pillars are not covered by NiO; the charge balance requires a halved pillar width and thus more demanding lithography. It is also unknown if the charge balance and high BV in such a hetero-SJ can be retained at high temperature.

This work addresses the above challenges via a novel self-aligned process, which allows for depositing the sidewall NiO and lifting off the cap NiO in a single, dry-etch-free step. This process obviates the need for the alignments to narrow pillars or adding the narrower pillars at device edge, and only a single GaN deep etch is needed. The fabricated GaN SJ diode shows a BV of 1.1 kV and a differential $R_{ON,SP}$ of $0.4 \text{ m}\Omega\cdot\text{cm}^2$. Both BV and $R_{ON,SP}$ show good thermal stability.

II. DEVICE DESIGN

Fig. 1(a) shows the schematic of the vertical GaN SJ diode. A GaN p-n junction is designed to avoid the surface breakdown and probe the true BV of GaN SJ; it can be a building block of future GaN SJ transistors. The epi consists of 40 nm p^{++} -GaN,

This work is supported in part by Office of Naval Research monitored by Lynn Petersen (Grant N00014-21-1-2183), National Science Foundation (Grant ECCS-2036740), and CPES Power Management Industry Consortium. Device fabrication was conducted as part of a user project at the Center for Nanophase Materials Sciences, which is a DOE Office of Science User Facility at Oak Ridge National Laboratory. (Corresponding author: Yuhao Zhang).

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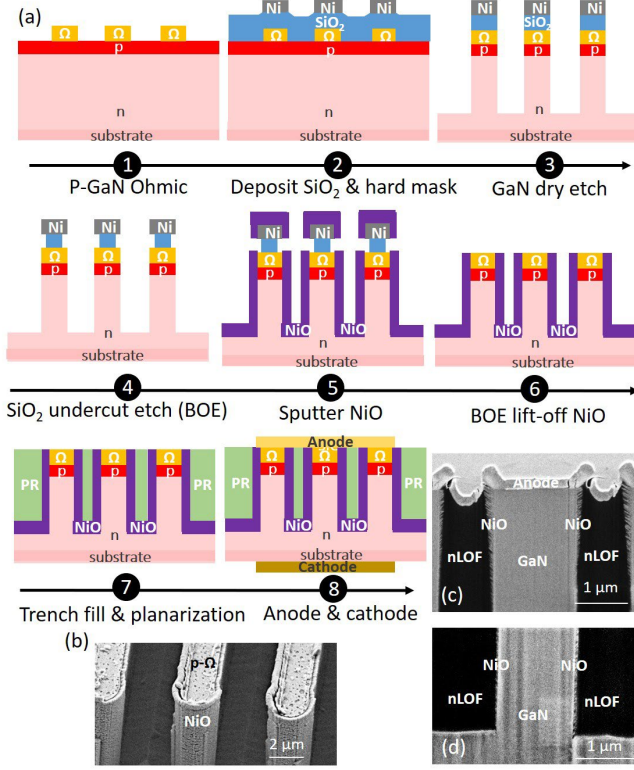


Fig. 2. (a) The main fabrication steps. (b) Top-view SEM images of the GaN pillars after the NiO cap lift-off. Cross-sectional SEM images of the final device with $w_n = 1.5 \mu\text{m}$ in the (c) top SJ and (d) pillar bottom regions.

300 nm p-GaN ([Mg]: 10^{19} cm^{-3}), 8 μm n-GaN ([Si]: $9 \times 10^{16} \text{ cm}^{-3}$), and 0.8 μm n⁺-GaIn, grown on 2-inch GaN substrates by Enkris Semiconductor Inc. Here n-GaN is doped $\sim 10\times$ higher than 1D devices. The sidewall NiO width (w_p) is designed based on $w_p N_A = N_D w_n / 2$ for charge balance, where w_n is the GaN pillar width and N_D is the donor concentration. The anode forms Ohmic contacts to both p-GaN and NiO.

The N_D and N_A are characterized with two test structures. The N_D is extracted to be $8 \times 10^{16} \text{ cm}^{-3}$ from the C-V data of a vertical GaN p⁺-n diode. The N_A of NiO can be tuned by the O₂ partial pressure in sputtering [16]. In this work, we use an Ar:O₂ flow rate of 20:1. The N_A of NiO is extracted to be $1.3 \times 10^{18} \text{ cm}^{-3}$ from the C-V data of a p-NiO/n⁺-Ga₂O₃ diode similar to [17]. When temperature increases from 25 °C to 125 °C, N_A shows an increase by 2.5%, while N_D shows nearly no change.

With the determined N_D and N_A , w_n is then optimized using the vertical SJ figure of merit ($FOM_{SJ} = BV/R_{on,sp}$). Here we set the GaN pillar spacing (S) to be 1 μm to avoid the NiO early coalescence issue in narrow trenches [16]. As shown in Fig. 1(b)-(c), the SJ's BV and $R_{on,sp}$ are modeled as a function of w_n . As w_n increases, the lateral E-field (E_x) increases, reducing the vertical E-field (E_y) and BV . Meanwhile, $R_{on,sp}$ decreases with the increased w_n due to a larger ratio (β) between the GaN conduction area and the total device area ($\beta = 1 + S/w_n$). Fig. 1(d) shows the modeled BV and FOM_{SJ} as a function of w_n , assuming that the SJ length (L_{SJ}) and n-GaN mobility (μ_n) are 6 μm and 600 cm^2/Vs , respectively. A w_n window of 1~3 μm is identified for the optimal FOM_{SJ} .

III. DEVICE FABRICATION AND SIMULATION

Fig. 2(a) shows the main steps of device fabrication. Ni/Au (5/20 nm) Ohmic contact to p-GaN is first formed with 560 °C annealing in the air. 1 μm thick SiO₂ is then deposited, followed by lifting off the Ti/Ni (10/340 nm) hard mask for GaN etching. The n-GaN pillar is etched with a $\sim 6.5 \mu\text{m}$ depth with the sidewall aligned to the m-plane, followed by a wet treatment to reduce the etch damage [18]. The sample is then dipped into 1:10 BOE for 40 s to produce an undercut on the SiO₂ sidewall. NiO is deposited by magnetron sputtering under the 60/3 sccm Ar/O₂ flow rate with other conditions identical to [17]. The undercut leaves the SiO₂ sidewall un-covered by NiO. A ~ 30 min rinse in 1:10 BOE follows to etch SiO₂ and naturally remove the NiO cap on GaN pillars.

A photoresist planarization process similar to [19] is then used to fill GaN trenches and connect all top contacts. The nLOF2020 is used, followed by a O₂ etch to expose the Ohmic metals. Following a flood exposure and baking, a Ni/Au metal pad is deposited as the anode, which also forms Ohmic contact to the sidewall NiO. The cathode is formed at the backside of GaN substrate. Devices with w_n of 1~3 μm are fabricated with S of 1 μm and a target w_p of $\sim 45 \text{ nm}$.

Fig. 2(b) shows the scanning electron microscopy (SEM) images of the device after the self-aligned process, revealing the complete lift-off of NiO caps and exposure of the p-GaN Ohmic metals. Fig. 2(c)-(d) shows the cross-sectional SEM images of the top and bottom SJ regions in the final device. The sidewall NiO thickness is found to decrease from $\sim 100 \text{ nm}$ to $\sim 40 \text{ nm}$ in the top 2 μm of the SJ region, and remains nearly constant in the bottom 4 μm SJ. This is due to the higher deposition rate of NiO near the planar surface. This higher planar deposition rate also leads to a thicker NiO at the trench bottom ($\sim 80 \text{ nm}$).

IV. DEVICE CHARACTERISTICS

Current density is normalized to the entire anode area. Fig. 3(a) shows the reverse I-V characteristics of the GaN SJ diodes with various w_n . The BV first increases with w_n , reaching 1100 V at w_n of 1.5 μm , and then decreases with w_n . The leakage current is below $5 \times 10^{-2} \text{ A/cm}^2$ before BV . Fig. 3(b) shows the box plots of BV as a function of the charge imbalance percentage (δ) in the main SJ region as calculated by $\delta = 1 - 2w_p N_A / w_n N_D$. Here w_p of 40 nm is used. For each δ , BV of six devices at different locations of the sample are measured. The result shows the strong impact of the SJ charge balance on BV , which is a key signature of SJ devices.

Fig. 3(c) shows the forward I-V characteristics of the device with $w_n = 1.5 \mu\text{m}$, revealing a differential $R_{on,sp}$ of 0.4 $\text{m}\Omega\text{-cm}^2$. Fig. 3(d) lists the $R_{on,sp}$ components. The p-GaN contact resistance accounts for 40% of $R_{on,sp}$. It is calculated by scaling the contact resistance measured by the transfer length method by the ratio β . The resistances of the p-GaN layer and the SJ drift region are ~ 0.03 and 0.13 $\text{m}\Omega\text{-cm}^2$, respectively, calculated using the respective layer thickness, β , and the carrier mobility and concentration.

Fig. 3(e) shows the reverse I-V characteristics at temperature (T) up to 125 °C. The non-destructive BV extracted at a 1 A/cm^2 compliance is nearly constant at 25~75 °C and starts to decrease at higher T , remaining $>1 \text{ kV}$ at 125 °C. The leakage current at 1 kV increases by only 3 times from 25 to 125 °C.

Fig. 3(f) shows the forward I-V characteristics at T up to 125

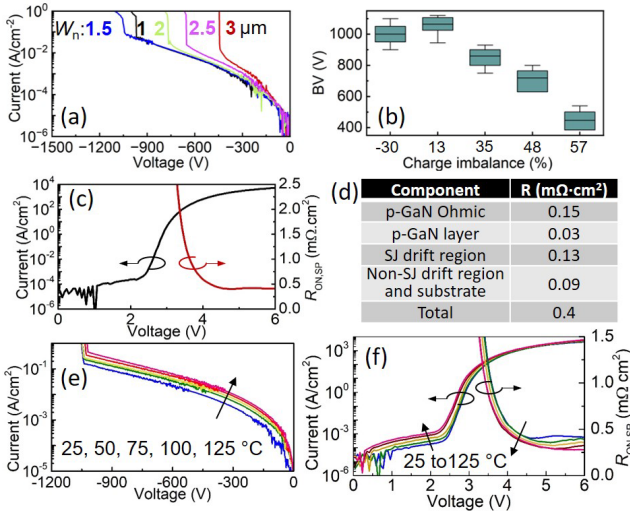


Fig. 3. (a) Reverse I-V characteristics of the vertical GaN SJ diodes with w_n of 1–3 μm and the identical S and w_p . (b) Box plot of the BV as a function of the SJ charge imbalance percentage. (c) Forward I-V characteristics and the extracted differential $R_{ON,SP}$ of the SJ diodes with $w_n=1.5 \mu\text{m}$. (d) Component of the total device $R_{ON,SP}$. Temperature-dependent (e) reverse I-V and (f) forward I-V and differential $R_{ON,SP}$ characteristics at 25 to 125 $^{\circ}\text{C}$ of the SJ diode with $w_n=1.5 \mu\text{m}$.

$^{\circ}\text{C}$, revealing a $R_{ON,SP}$ drop to $0.3 \text{ m}\Omega\cdot\text{cm}^2$. The $R_{ON,SP}$ reduction at high T could be due to two mechanisms: a) reduced p-GaN resistance and Ohmic contact resistance [20], and b) enhanced conduction via the p-NiO/n-GaN path due to the reduced NiO resistivity. The first mechanism is proved by the $R_{ON,SP}$ of 1D GaN p-n diodes fabricated on the same wafer, which decreases from $0.28 \text{ m}\Omega\cdot\text{cm}^2$ to $0.25 \text{ m}\Omega\cdot\text{cm}^2$ from 25 $^{\circ}\text{C}$ to 125 $^{\circ}\text{C}$. The latter mechanism is supported by current increase with T in the SJ diode at the forward voltage below 2 V. In this bias range, minimal current increase is observed in the GaN p-n diode (the built-in potential of the NiO/GaN and GaN p-n junction is $\sim 1.6 \text{ V}$ and 3 V , respectively).

Simulation is used to understand the breakdown mechanism of the fabricated SJ device considering the non-uniform NiO thickness at the trench bottom and sidewall. In addition to the global SJ charge balance, the other determining factor of device BV is the local E-field crowding at the trench bottom formed by the thicker NiO. For $w_n = 1.5 \mu\text{m}$, the peak E-field is located at the trench bottom (Fig. 4(a)). As this E-field is induced by the locally excessive p-type charge, it is sensitive to N_A . At high T , the higher N_A , despite possibly improving the global SJ charge balance, leads to higher E-field stress at the trench bottom (Fig. 4(b)). This could explain the negative temperature coefficient of the experimental device BV .

This factor also impacts the w_n modulation of device BV . At $w_n = 1 \mu\text{m}$, the peak E-field is still located at the trench bottom (Fig. 4(c)), suggesting its dominant role on BV . At $w_n = 2 \mu\text{m}$, the excessive n-type charge moves the peak E-field to the NiO/GaN vertical sidewall at the top SJ region (Fig. 4(d)), as the impact of the global SJ charge imbalance becomes dominant. This can explain the BV 's strong sensitivity on charge imbalance at $w_n > 1.5 \mu\text{m}$ [see Fig. 3(b)].

Fig. 5 compares the differential $R_{ON,SP}$ versus BV trade-off of the vertical GaN SJ diode with the state-of-the-art vertical GaN 1D diodes [21]–[30] and SiC SJ devices [4]–[6], as well as the

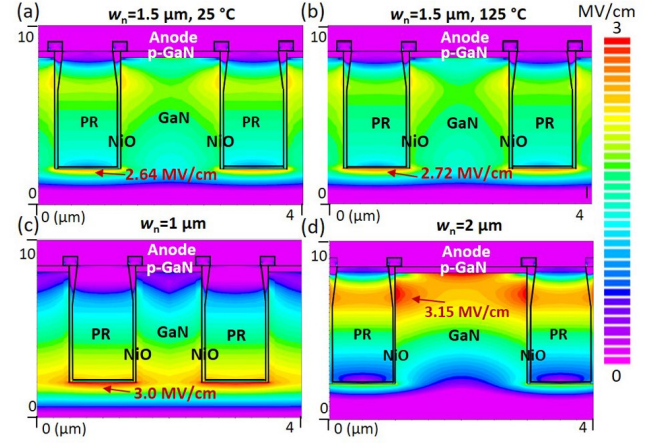


Fig. 4. Simulated E-field contours in the SJ diodes with (a) $w_n = 1.5 \mu\text{m}$, (b) the same w_n and 2.5% higher N_A , (c) $w_n = 1 \mu\text{m}$, and (d) $w_n = 2 \mu\text{m}$ at 1.1 kV.

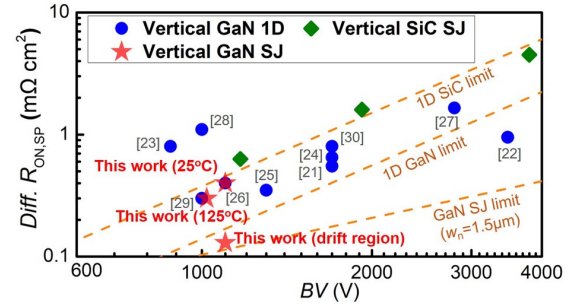


Fig. 5. Differential $R_{ON,SP}$ versus BV trade-off of the vertical GaN SJ diode in comparison with vertical GaN diodes and SiC SJ devices. The high-temperature performance and drift-region resistance of our device are also marked. In the 1D limit, a critical E-field of 3 MV/cm and 2.5 MV/cm [31] and mobility of $1200 \text{ cm}^2/\text{Vs}$ and $800 \text{ cm}^2/\text{Vs}$ are used for GaN and SiC, respectively. The GaN SJ limit assumes a higher critical E-field of 3.3 MV/cm (due to higher doping) and a mobility of $600 \text{ cm}^2/\text{Vs}$.

1D SiC/GaN limits and the GaN SJ limit [$R_{on,sp} = 4w_nBV/(\epsilon\mu_n E_c^2)$]. The performance of our device is among the best in vertical GaN diodes with a similar BV . The device performance at 125 $^{\circ}\text{C}$ shows little degradation. The trade-off between BV and the drift region resistance is superior to the 1D GaN limit, fulfilling the theoretical promise of SJ devices in GaN. To fully exploit the SJ superiority, it is desirable to continue upscaling the BV as a) the SJ drift region will contribute an increasing percentage of device $R_{ON,SP}$, and b) the SJ theoretical limit will increasingly outperform the 1D limit.

V. SUMMARY

We present the vertical GaN SJ diode fabricated by a novel self-aligned process to deposit NiO at the GaN pillar sidewalls. The device achieves a BV over 1 kV and $R_{ON,SP}$ of $0.4 \text{ m}\Omega\cdot\text{cm}^2$, the trade-off of which is among the best in SJ devices. At 125 $^{\circ}\text{C}$, BV maintains over 1 kV and $R_{ON,SP}$ drops to $0.3 \text{ m}\Omega\cdot\text{cm}^2$. To date, such high temperature operation is reported in a hetero-SJ device for the first time. These results show the promise of GaN SJ power devices, and the self-signed process has the wide material and device applicability.

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