

# High-pressure Synthesis of Cobalt Polynitrides: Unveiling Intriguing Crystal Structures and Nitridation Behavior

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**Abstract** In this study, we conduct extensive high-pressure experiments to investigate phase stability in the cobalt-nitrogen system. Through a combination of synthesis in a laser-heated diamond anvil cell, first-principles calculations, Raman spectroscopy, and single-crystal X-ray diffraction, we establish the stability fields of known high-pressure phases, hexagonal NiAs-type CoN, and marcasite-type CoN<sub>2</sub> within the pressure range of 50-90 GPa. We synthesize and characterize previously unknown nitrides, Co<sub>3</sub>N<sub>2</sub>, *Pnma*-CoN and two polynitrides, CoN<sub>3</sub> and CoN<sub>5</sub>, within the pressure range of 90-120 GPa. Both polynitrides exhibit novel types of polymeric nitrogen chains and networks. CoN<sub>3</sub> feature branched-type nitrogen trimers (N<sub>3</sub>) and CoN<sub>5</sub> show  $\pi$ -bonded nitrogen chain. As the nitrogen content in the cobalt nitride increases, the CoN<sub>6</sub> polyhedral frameworks transit from face-sharing (in CoN) to edge-sharing (in CoN<sub>2</sub> and CoN<sub>3</sub>), and finally to isolated (in CoN<sub>5</sub>). Our study provides insights into the intricate interplay between structure evolution, bonding arrangements, and high-pressure synthesis in polynitrides, expanding the knowledge for the development of advanced energy materials

## Introduction

Polynitrides are a class of compounds containing structural units, exhibiting nitrogen-nitrogen bonds, including azides, diazenides, pentazoles, etc.<sup>[1-3]</sup> These compounds have recently undergone extensive investigation through high-pressure experiments, where they were synthesized in laser-heated diamond anvil cells.<sup>[1-7]</sup> Transition metal pernitrides (e.g. PtN<sub>2</sub>, ReN<sub>2</sub>, OsN<sub>2</sub>) containing N-N dimers are recognized for their high bulk modulus and potential applications as superhard materials.<sup>[1,8,9]</sup> Polynitrides containing polymeric nitrogen chains (such as FeN<sub>4</sub>,<sup>[2]</sup> TaN<sub>5</sub>,<sup>[10]</sup> MgN<sub>4</sub>,<sup>[11]</sup> WN<sub>10</sub>,<sup>[12]</sup> YN<sub>6</sub>,<sup>[13]</sup>) represent higher levels of nitridation. More recently, experimental studies have uncovered nitrogen rings including pentazoles (N<sub>5</sub><sup>-</sup>), hexazine anions N<sub>6</sub><sup>x-</sup> and N<sub>18</sub>.<sup>[3,13,14,15]</sup>

The presence of dimers, chains, and rings in polynitrides can contribute to their potential usage as high-energy density material,<sup>[15,16]</sup> because converting singly or doubly bonded nitrogen to triple-bonded nitrogen can release a substantial

amount of energy. Therefore, there has been an active pursuit of the synthesis and prediction of polynitrides as prospective high-energy density materials.<sup>[2,3,6,16]</sup>

Cobalt nitrides have emerged as promising candidates for potential polynitride with higher energy density.<sup>[17]</sup> Several cobalt nitrides were experimentally identified, such as  $\text{Co}_2\text{N}$ ,<sup>[18,19]</sup>  $\text{Co}_3\text{N}$ ,<sup>[20]</sup>  $\text{Co}_4\text{N}$ ,<sup>[20]</sup>  $\text{CoN}$ ,<sup>[5]</sup> and  $\text{CoN}_2$ .<sup>[5]</sup> The cobalt nitrogen system had been investigated in comparison with the iron nitrogen system.<sup>[5]</sup> These two chemical systems share similarities, as both iron and cobalt pernitride exhibit a marcasite-type structure.<sup>[2,5,21]</sup> Additionally, both FeN and CoN undergo a transition from a ZnS-type structure to a NiAs-type structure at higher pressures.<sup>[5]</sup> Iron nitrides start to form polynitride anions  $[\text{N}_4^{2-}]_n$  at 106 GPa.<sup>[2]</sup> However, it is currently unknown whether similar behavior occurred in cobalt nitrides at high pressure, approximately 100 GPa.

Understanding the nitridation path for transition metals might greatly help develop nitrogen-based materials at high pressure.<sup>[21,22]</sup> While there was not a single universal nitridation path applicable to all transition metals, first-principle calculation reported the progressive nitridation trend for transition metals.<sup>[17,23]</sup> Numerous experimental studies had demonstrated a wealth of polymorphs of transition metal nitrides at different pressure and temperature conditions.<sup>[4,6,21,22]</sup> However, to uncover the nitridation path, the nitridation behavior of transition metals at high-pressure needs to be explored.

## Results and Discussion

We conducted the synthesis of cobalt nitride starting from metallic cobalt and nitrogen in diamond anvil cells. In our typical experimental procedure, we initially compressed the sample in the diamond anvil cell to the desired pressure. Subsequently, we employed an infrared laser, coupled with cobalt metal to heat the sample to temperatures ranging from 1700–2200 K. We investigated their structure and composition using synchrotron X-ray diffraction (XRD) and Raman spectroscopy. The experiments are summarized in Table 1.

Laser heating at pressures between 50–90 GPa resulted in the formation of two known high-pressure phases: hexagonal NiAs-type CoN and marcasite-type  $\text{CoN}_2$ <sup>[5]</sup> (Figure 1a, 1d, and S1). Additionally, we identified  $\text{Pnma-Co}_3\text{N}_2$ ,  $\text{Pnma-CoN}$ ,  $\text{CoN}_3$ , and  $\text{CoN}_5$  in the Co–N system from 90–120 GPa (Figures 1b, 1c, 1e, 1f, S2, and S3). To obtain more comprehensive and detailed information on these crystals, we employed single-crystal XRD techniques (see Supporting information) together with powder diffraction and Raman spectroscopy to identify the crystal structures and composition of all phases.

**Table 1.** Experimental runs

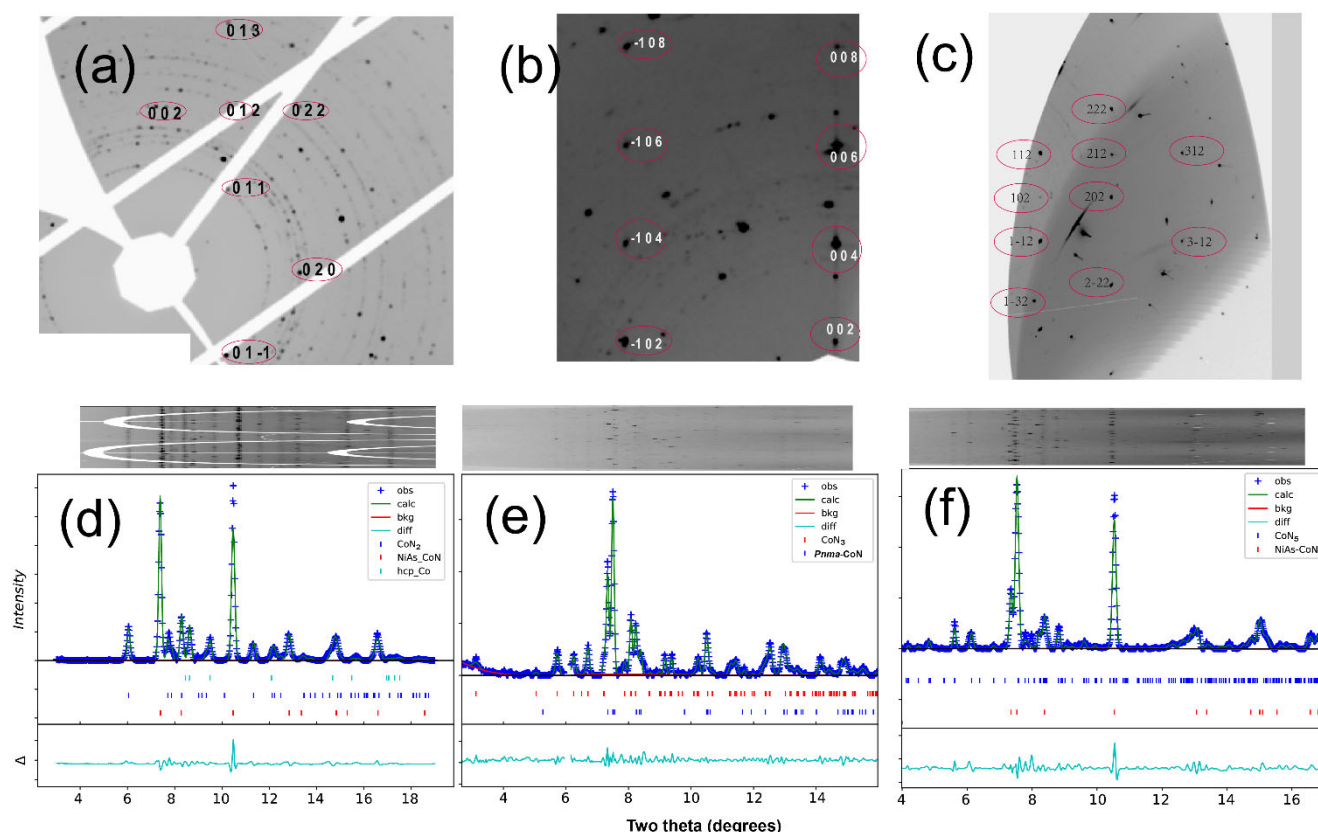
P (GPa) <sup>[a]</sup>	T (K) <sup>[b]</sup>	Beamline	Results	Structures and Composition <sup>[c]</sup>
97(5)	2200(200)	Petra III	$\text{CoN}_3 + \text{CoN}_5 + \text{CoN} + \text{Co}_3\text{N}_2$	$\text{Pnma-CoN}_3$ $\text{C2/c-CoN}_5$
58(3)	1850(150)	GSECARS	$\text{CoN}_2 + \text{CoN}$	$\text{P6}_3/\text{mmc-CoN}$
67(4)	1760(200)	GSECARS	$\text{CoN}_2 + \text{CoN}$	$\text{Pnnm-CoN}_2$
117(6)	2460(300)	Petra III	$\text{CoN} + \text{Co}_3\text{N}_2 + \text{CoN}_3 + \text{CoN}_5$	$\text{Pnma-CoN}$ $\text{Pnma-Co}_3\text{N}_2$

[a] P (Pressure), [b] T (Temperature), Starting materials: Cobalt and Nitrogen.

[c] We included the crystal structures deposited in CCDC (Cambridge Crystallographic Data Centre).<sup>[24]</sup> Deposition numbers are CSD 2325509 (for  $\text{Pnma-Co}_3\text{N}_2$ ), 2325510 ( $\text{P6}_3/\text{mmc-CoN}$ ), 2325511 (for  $\text{Pnma-CoN}$ ), 2308801 (Marcasite-type  $\text{CoN}_2$ ), 2325513 (for  $\text{Pnma-CoN}_3$ ), 2325514 (for  $\text{C2c-CoN}_5$ ).

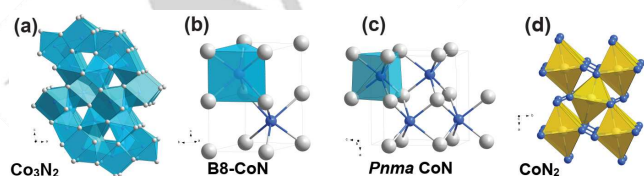
The crystal structure of NiAs-type CoN is shown in Figure 2b, which can be described as a hexagonal cell with the space group  $\text{P6}_3/\text{mmc}$ . At a pressure of 55 GPa, the unit cell parameters are  $a = 2.726(2)$  Å and  $c = 4.713(2)$  Å. Detailed crystal information from single-crystal diffraction can be found in Table S1. In the NiAs-type CoN structure, the coordination number for cobalt is 6. The  $\text{CoN}_6$  octahedra share faces in the NiAs-type structure. There is no close N–N bonding (typical bond lengths  $<1.5$  Å) observed in the NiAs-type CoN. At higher pressure (110 GPa), we found  $\text{Pnma}$ -type CoN (Figure 2c), which is a distorted NiAs-type structure (see Table S2). The  $\text{Pnma}$ -type structure was documented in FeN as the higher pressure form.<sup>[23]</sup>

The marcasite-type structure of  $\text{CoN}_2$ , initially reported using powder diffraction<sup>[5]</sup>, is depicted in Figure 2d. The peaks observed in single-crystal diffraction can be indexed to the  $\text{Pnnm}$  space group (Figure 1a). At a pressure of 67 GPa, the unit cell parameters are determined as  $a = 4.307(6)$  Å,  $b = 3.597(3)$  Å, and  $c = 2.445(8)$  Å. Atomic parameters for  $\text{CoN}_2$  can be found in Table S3. The crystal structure of  $\text{CoN}_2$ , shown in Figure 2d, exhibits a marcasite-type structure within the pressure range of 50–90 GPa. In this structure, the coordination number of cobalt is 6. Edge-sharing octahedra form infinite chains along the  $z$ -axis. Two adjacent octahedra between two chains are closely positioned. The nearest nitrogen–nitrogen distance in  $\text{CoN}_2$  is  $1.27(2)$  Å, which indicates a double nitrogen–nitrogen bond, as is known in diazenides.<sup>[25,26]</sup>



**Figure 1.** XRD data in the Co-N system at high pressures. (a) Indexation of marcasite-type  $\text{CoN}_2$  from the unwarped single-crystal diffraction image at 67 GPa. (b) Indexation of  $\text{CoN}_3$  from the unwarped single-crystal diffraction images at 120 GPa. (c) Indexation of  $\text{CoN}_5$  from the unwarped single-crystal diffraction images at 120 GPa. (d) Rietveld refinements of powder diffraction patterns collected at 67 GPa, Calculated diffraction lines for the marcasite-type  $\text{CoN}_2$  (blue ticks), NiAs-type CoN (red ticks), and  $\text{hcp-Co}$  are shown in figures. The X-ray wavelength  $\lambda=0.2952$  Å, residue for refinement  $R_{\text{wp}}=9\%$  (e) Rietveld refinements of powder diffraction patterns collected at 120 GPa, calculated diffraction lines for  $Pnma$ -type CoN (blue ticks) and  $\text{CoN}_3$  (red ticks) are exhibited in the figure, The X-ray wavelength  $\lambda=0.2905$  Å,  $R_{\text{wp}}=12\%$ . (f) Rietveld refinements of powder diffraction patterns collected at 120 GPa, calculated diffraction lines for NiAs-type CoN (red ticks) and  $\text{CoN}_5$  (blue ticks) are exhibited in the figure, X-ray wavelength  $\lambda=0.2900$  Å,  $R_{\text{wp}}=15\%$ . The Miller index for single crystal diffraction spots is shown in red circles in (a, b, and c) Unrolled projection for 2D powder diffraction images were shown on top of d, e and f.

We discovered  $\text{Co}_3\text{N}_2$  at a pressure range from 90-120 GPa. The  $\text{Co}_3\text{N}_2$  has an orthorhombic structure with the space group  $Pnma$ . The lattice parameters at 120 GPa for  $\text{Co}_3\text{N}_2$  are  $a=5.228(2)$  Å,  $b=2.523(5)$  Å and  $c=9.972(5)$  Å (Table S4). The cobalt atoms in  $\text{Co}_3\text{N}_2$  have a coordination number of 4-5. A similar  $\text{Fe}_3\text{N}_2$  had been documented in Fe-N system.<sup>[2]</sup>

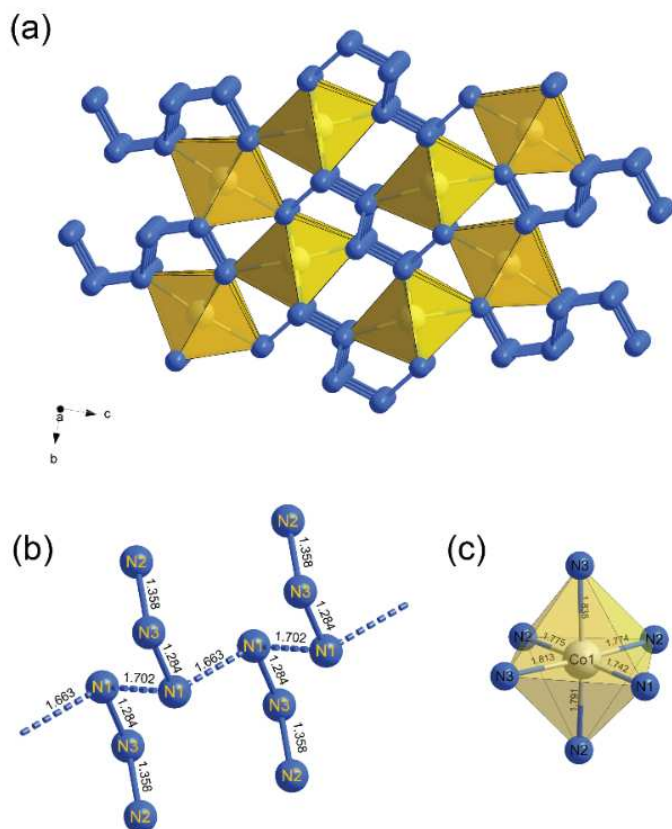


**Figure 2.** Crystal structures of  $\text{Co}_3\text{N}_2$  (a), NiAs-type (B8) CoN (b),  $Pnma$ -CoN (c), marcasite-type  $\text{CoN}_2$  (d). Yellow polyhedra represents Co-N framework. Blue polyhedral represent N-Co framework. The closely bonded nitrogen atoms are connected by blue lines in (d).

We discover two new cobalt polynitride compounds  $\text{CoN}_3$  and  $\text{CoN}_5$ , within the pressure range of 90-120 GPa, the structures of which do not correspond to any existing cobalt nitride or known nitrogen phases at high pressure (Figures 3 and 4). The crystal

structure of the new compound  $\text{CoN}_3$  has an orthorhombic symmetry (space group  $Pnma$ ), with the following unit cell parameters:  $a = 2.555(7)$  Å,  $b = 3.463(2)$  Å, and  $c = 10.655(5)$  Å at 100 GPa. Structure solution and refinement revealed the chemical formula  $\text{CoN}_3$  (detailed crystal structure information is provided in the Table S5). Based on 2D- XRD images (Fig. 1b), the structure might contain certain amount of structure disorder at high pressure. One symmetry-independent Co as well as three N atoms occupy the Wyckoff sites 4e. Similar to  $\text{CoN}_2$ , the cobalt atom is octahedrally coordinated by six nitrogen atoms in a slightly distorted  $\text{CoN}_6$  octahedron (Figure 3). The  $\text{CoN}_6$  octahedra are interconnected by sharing edges along the  $a$ -axis, forming infinite chains. These chains are interconnected through vertices forming a zig-zag pattern along the  $b$ -axis. Nitrogen atoms form an unprecedented arrangement (Figure 3). N1, N2, N3 atoms form bent  $\text{N}_3$  units ( $\text{N2-N3-N1}$  angle of  $110.5(9)^\circ$ ). These  $\text{N}_3$  units are interconnected by longer N1-N1 bonds into infinite chains along the  $a$ -axis. The bond distances in the  $\text{N}_3$  units are  $1.284(17)$  Å ( $\text{N1-N3}$ ) and  $1.358(13)$  Å ( $\text{N3-N2}$ ), which are typical for covalent N-N bonds. The inter- $\text{N}_3$  distances of  $1.66(3)$ - $1.70(2)$  Å are significantly longer than typical covalently bonded nitrogen atoms in polynitrides, manifesting that  $\text{CoN}_3$  is a unique compound

featuring the intermediate state between a polynitride with oligomeric units (bent  $N_3$ ) and a polymeric branched chain (like in  $TaN_5$ <sup>[10]</sup>). The calculated isosurfaces of electron localization function (ELF) in Figure S8, which support the N3-N3 bonding at 112 GPa. However, the unusual N-N bonding will disappear during decompression.

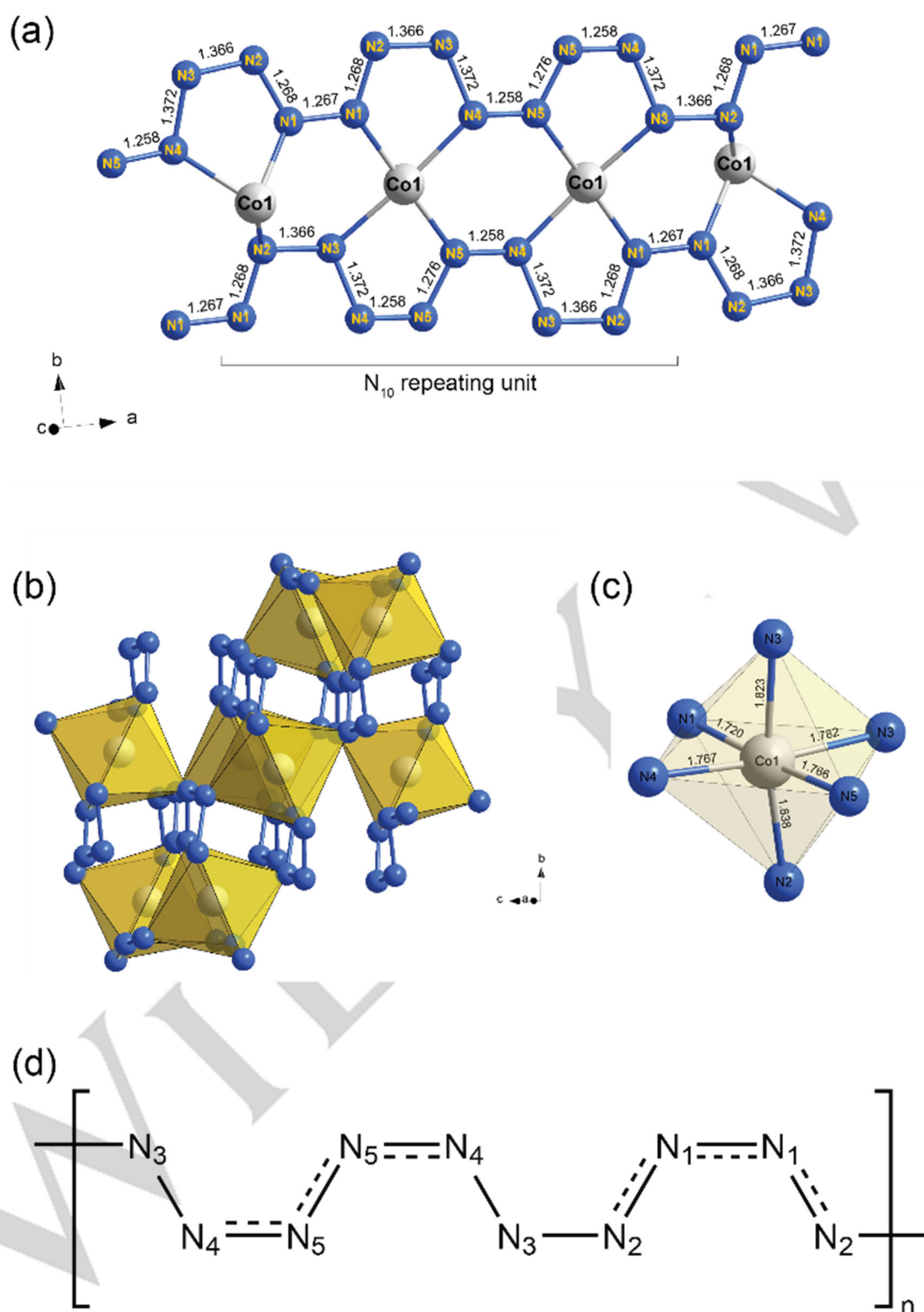


**Figure 3.** Crystal structure of  $CoN_3$  at 100 GPa.  $CoN_6$  polyhedra are shown by yellow octahedra in (a) and (c). N-N covalent bonds are depicted by solid blue lines while longer N-N bonds are shown by dotted lines in (b).

The crystal structure of  $\text{CoN}_5$  has a monoclinic symmetry (space group  $C2/c$ ), with the unit cell parameters:  $a=4.951(5)$  Å,  $b=6.910(8)$  Å,  $c=8.062(6)$  Å,  $\beta=95.77(5)^\circ$ . Single-crystal diffraction is used to determine the crystal structure (Table S6). In the crystal structure (Figure 4), a single symmetry independent Co atom and five nitrogen atoms occupy Wyckoff sites  $8f$ . The cobalt atom maintains its six-fold coordination, located in the distorted  $\text{CoN}_6$  octahedra (Figure 4d). These octahedra do not share common vertices, edges or faces, but are interconnected through polymeric nitrogen chains (Figure 2e). Despite the apparent simplicity of the non-branched polymeric nitrogen chain, crystal chemical analysis shows that the shortest repeating unit consists of 10 nitrogen atoms [-N1-N2-N3-N4-N5-N5-N4-N3-N2-

N1-] (Figure 4d). Within the chains, all atoms except N3 have three-fold coordination by one Co and two N atoms, manifesting  $sp^2$  hybridization. The N3 atom is tetrahedrally coordinated by two Co and two N atoms, suggesting  $sp^3$  hybridization of these atoms. Therefore, N3-N2 and N3-N4 bonds are of single character, which is in agreement with refined interatomic distances (1.366(2) and 1.372(4) Å, respectively). These distances are in good agreement with those in single-bonded N polymorphs (*bp-N* and *cg-N*).<sup>[27,28]</sup> The  $sp^2$  hybridization of N1, N2, N4 and N5 atoms suggest that there exist an extended  $\pi$ -system in *cis*- and *trans*-tetrazene fragments (N4-N5-N5-N4 and N2-N1-N1-N2, respectively (Figure 4d)).





**Figure 4.** Crystal structure of CoN<sub>5</sub> at 100 GPa; (a) shows atomic arrangements of CoN<sub>5</sub>; (b) shows the crystal structure of CoN<sub>5</sub>; (c) shows the CoN<sub>5</sub> octahedra; (d) shows the repeated units of N<sub>10</sub> fragments in CoN<sub>5</sub>.

In our powder X-ray diffraction experiments, we consistently observe the co-existence of  $\text{CoN}+\text{CoN}_2$  phases in the X-ray diffraction patterns within the pressure range of 50-90 GPa (Figures 1d and S1). Similarly, we observe the coexistence of  $\text{Co}_3\text{N}_2+\text{CoN}+\text{CoN}_3+\text{CoN}_5$  in the diffraction patterns within the pressure range of 90-120 GPa (Figures 1 and S2). These observations indicate that these phases are close to thermodynamic stability within their respective pressure ranges as it will be discussed below. Specifically, NiAs-type  $\text{CoN}$  and  $\text{CoN}_2$  are stable from 50-90 GPa,  $\text{Co}_3\text{N}_2+\text{Pnma-CoN}+\text{CoN}_3+\text{CoN}_5$  phases are stable from 90-120 GPa. The coexistence of multiple stoichiometries of cobalt nitride at the same pressure can be due to a possible variation of nitrogen content in the diamond anvil cell cavity during the high-temperature synthesis. Multiple phases (with a low formation energy) that are stable or metastable form the phase assemblages on temperature quenching at nominally the same pressure. We calculated the unit cell volumes of all the phases per formula in comparison with the atomic volumes of cobalt and polymeric nitrogen (Figure S4). The agreements observed between these values support our crystal structure solutions including  $\text{Co}_3\text{N}_2$ ,  $\text{CoN}$ ,  $\text{CoN}_2$ ,  $\text{CoN}_3$ , and  $\text{CoN}_5$ . It is worthwhile to note that  $\text{CoN}_3$  and  $\text{CoN}_5$  are not quenchable to ambient conditions (determined by Raman spectroscopy on unloading at 300 K) while NiAs-type  $\text{CoN}$  and  $\text{CoN}_2$  can be recovered at ambient condition.<sup>[5]</sup>

In our research, we conducted Raman spectroscopy measurements of cobalt nitride samples under high-pressure conditions. At a pressure of 80 GPa, we identify the Raman peaks at approximately 900 and 1280  $\text{cm}^{-1}$  from the heated spots on the sample. These Raman frequencies do not correspond to any known molecular nitrogen Raman vibrational modes, which have lower frequencies (Figure 5b). To gain further insights into the observed Raman mode, we compare Raman spectra from this study with the Raman spectra of marcasite-type  $\text{FeN}_2$ .<sup>[21]</sup> The marcasite-type  $\text{FeN}_2$  exhibit similar Raman peaks at 740 and 1100  $\text{cm}^{-1}$ .<sup>[21]</sup> These peaks correspond to the symmetric

deformation and stretching modes ( $A_g$ ) within the marcasite-type structure.<sup>[21,29]</sup> Notably, previous studies<sup>[5]</sup> did not observe any Raman signals in recovered samples of marcasite-type  $\text{CoN}_2$ . This is because the Raman spectra of  $\text{CoN}_2$  are weak, as this compound is likely metallic based on the results of our electronic band structure calculations (Figure S5).

After laser heating the sample at 120 GPa, we observe weak Raman peaks, which could be tentatively assigned to  $\text{CoN}_3+\text{CoN}_5$  (Figure 5a). These peaks are distinguished from the Raman signal from molecular  $\theta\text{-N}_2$ <sup>[30]</sup> as well as polymeric *cg-N* and *bp-N*.<sup>[27,28,31]</sup> Group theory predicts 24 Raman modes ( $8A_{1g}+4B_{1g}+8B_{2g}+4B_{3g}$ ) for the *Pnma*- $\text{CoN}_3$  structure, whereas 36 Raman active modes ( $18A_g+18B_g$ ) are expected in *C2/c*- $\text{CoN}_5$ . We also theoretically calculated the Raman modes for  $\text{CoN}_3$  and  $\text{CoN}_5$ . We note that the Raman modes of  $\text{CoN}_3+\text{CoN}_5$  can be divided into three major groups separated in the frequency ranges. The high-frequency group (two or more modes) correspond to the N-N stretching vibrations; their frequencies are consistent with single- and double bonded N-N atoms. The middle-frequency range (three modes at 650-800  $\text{cm}^{-1}$ ) corresponds to deformation modes of  $\text{N}_3$  or  $\text{N}_5$  units. The low-frequency range (one mode) corresponds to vibrations of  $\text{N}_3$  or  $\text{N}_5$  units (weakly bonded) with respect to each other. Due to the complexity of multiple phases ( $\text{CoN}+\text{CoN}_3+\text{CoN}_5$ ), we are not able to assign all Raman modes (Figure 5). Most of the peaks can be explained by  $\text{CoN}_5$  (Table S7). The differences in frequency between calculated and experimental Raman modes are within 10%, with most of them are being extremely close (less than 3%) (Table S7). The typical Raman shifts differences from calculations and experimental results for strong correlated electronic systems are about 10%.<sup>[32,33]</sup> The good fitness of calculated Raman suggests that our sample contains polynitride  $\text{CoN}_5$  and  $\text{CoN}_3$ . Due to the metallicity of  $\text{CoN}_3$  and  $\text{CoN}_5$  (Figures. S6, S7), some of the predicted Raman peaks could not be observed in the spectra because they are too weak. In sum, our Raman spectra support the existence of polynitride compounds in Co-N system at 90-120 GPa.

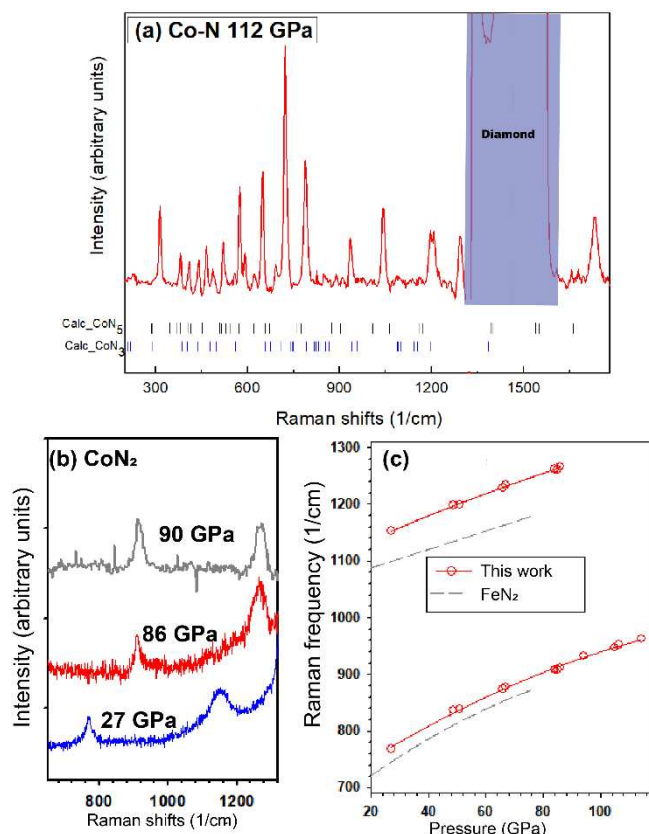


Figure 5. Raman spectra of  $\text{CoN}_3+\text{CoN}_5$  at 120 GPa (a) and marcasite-type  $\text{CoN}_2$  (b). (c) Pressure dependences of the Raman frequencies for the marcasite-type  $\text{CoN}_2$  together with Raman data from  $\text{FeN}_2$ .<sup>[21]</sup>

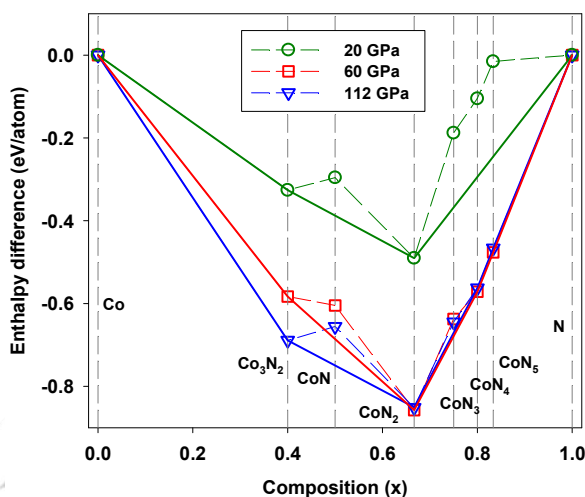


Figure 6. Convex hull diagram calculated from density functional theory of cobalt nitride systems at 20, 60, and 112 GPa. Stable phase include NiAs-type  $\text{CoN}$ ,  $\text{CoN}_2$ ,  $\text{CoN}_3$ ,  $\text{CoN}_4$ <sup>[2]</sup> and  $\text{CoN}_5$  (black triangle). The stable phases of nitrogen are assumed to be  $\epsilon\text{-N}_2$  at 20 and 60 GPa and  $\text{cg-N}$  – at 112 GPa.

We calculated the convex hull diagram for Co-N nitrides (see the Supplementary Information) at 20, 60, and 112 GPa (Figure 6). The stability fields of cobalt nitride phases were determined by calculating the formation enthalpies at the pressure range 20–112 GPa. The points connected by the solid lines in Figure 6 correspond to the thermodynamically stable phases. We find that  $\text{CoN}$ ,  $\text{CoN}_2$ , and  $\text{Co}_3\text{N}_2$  are thermodynamically stable or metastable compounds at 20 GPa. At 60 and 112 GPa,  $\text{CoN}_4$  and  $\text{CoN}_5$  gradually become stable, and  $\text{CoN}_3$  stays very close to

thermodynamic stability line. The crystal structures obtained from calculations and experimental studies are consistent. We did not observe  $\text{CoN}_4$ , which is expected to form infinite nitrogen chains<sup>[2]</sup> similar to  $\text{CoN}_5$ .

## Conclusion

The discovery of  $\text{CoN}_3$  and  $\text{CoN}_5$  phases represents a significant advancement in our understanding of polynitride structures.  $\text{CoN}_3$



is observed where Co atoms are connected by nitrogen trimer bonds. This branched-type structure arrangement in  $\text{CoN}_3$  can be considered as an intermediate state between octahedra connected by N-N bonds and the formation of branched or infinite -N-N- chains. At extremely high pressures, metal nitrides tend to form long chains or even ring structures. In this context,  $\text{CoN}_5$  shows an infinite  $\pi$  bonded nitrogen chain at higher pressure in its crystal structure. Progressive nitridation is clearly observed in our study from  $\text{CoN}$ ,  $\text{CoN}_2$  to  $\text{CoN}_3$ , and  $\text{CoN}_5$  with an increasing degree of nitrogen polymerization.

Another intriguing finding of our study is the correlation between  $\text{CoN}_6$  octahedra and the increasing nitrogen content in the unit cell. The coordination number is based on the ionic radii between cation and anion in ionic crystals.<sup>[34]</sup> In the cobalt nitrogen system, Co atoms are preferentially coordinated with six nitrogen atoms within the pressure range of 50–120 GPa. Polynitrogen anions influence  $\text{CoN}_6$  octahedral connectivity in these structures, leading to the formation of face sharing ( $\text{CoN}$ ) to edge sharing ( $\text{CoN}_2$  and  $\text{CoN}_3$ ) octahedra. In order to accommodate more nitrogen in the crystal structure, the  $\text{CoN}_6$  octahedra in  $\text{CoN}_5$  become disconnected, leaving enough space for the polymerized nitrogen atoms. This finding adds to our understanding of the structure evolution and bonding behaviors of polynitrides under various pressure conditions, shedding light on the intricate interplay between  $\text{CoN}_6$  octahedra arrangements and nitrogen polymerization of polynitride.

In sum, we experimentally and theoretically explored the cobalt nitride system at pressure up to 120 GPa. With the single-crystal XRD, we successfully solved several phases including NiAs-type  $\text{CoN}$ , marcasite-type  $\text{CoN}_2$ ,  $Pnma$ - $\text{CoN}$ ,  $\text{Co}_3\text{N}_2$ , and the novel polynitrides  $\text{CoN}_3$  and  $\text{CoN}_5$ .  $\text{CoN}_3$  and  $\text{CoN}_5$  are very different from iron polynitride  $\text{FeN}_4$  observed in the Fe-N system. These observations emphasize the need for additional experimental studies and first-principles calculations to gain a deeper understanding of the nitridation processes and to facilitate the design of transition metal polynitride materials.

## Supporting Information

The authors have cited additional references within the Supporting Information.<sup>[35–53]</sup> Experimental details, additional data are provided in the Supporting Information.

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## Conflict of Interest

The authors declare no conflict of interest.

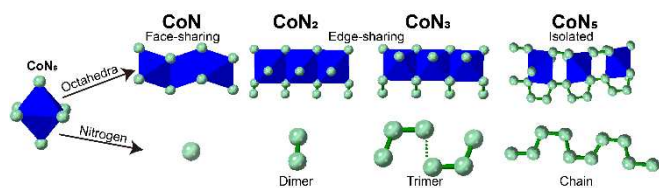
**Keywords:** Polynitride • Cobalt • Crystal Structure • High-Pressure • Diamond-Anvil Cell.

## References used

- [1] J. C. Crowhurst, A. F. Goncharov, B. Sadigh, C. L. Evans, P. G. Morrall, J. L. Ferreira, A. J. Nelson, *Science* **2006**, *311*, 1275–1278.
- [2] M. Bykov, E. Bykova, G. Aprilis, K. Glazyrin, E. Koemets, I. Chuvashova, I. Kuppenko, C. McCammon, M. Mezouar, V. Prakapenka, H.-P. Liermann, F. Tasnádi, A. V. Ponomareva, I. A. Abrikosov, N. Dubrovinskaia, L. Dubrovinsky, *Nat Commun* **2018**, *9*, 2756.
- [3] Y. Wang, M. Bykov, I. Chepkasov, A. Samtsevich, E. Bykova, X. Zhang, S. Jiang, E. Greenberg, S. Chariton, V. B. Prakapenka, A. R. Oganov, A. F. Goncharov, *Nat. Chem.* **2022**, *14*, 794–800.
- [4] T. Sasaki, T. Yamamoto, S. Asano, K. Niwa, M. Hasegawa, *Dalton Trans.* **2023**, *52*, 469–475.
- [5] K. Niwa, T. Terabe, D. Kato, S. Takayama, M. Kato, K. Soda, M. Hasegawa, *Inorg. Chem.* **2017**, *56*, 6410–6418.
- [6] M. Bykov, E. Bykova, E. Koemets, T. Fedotenko, G. Aprilis, K. Glazyrin, H.-P. Liermann, A. V. Ponomareva, J. Tidholm, F. Tasnádi, I. A. Abrikosov, N. Dubrovinskaia, L. Dubrovinsky, *Angewandte Chemie International Edition* **2018**, *57*, 9048–9053.
- [7] H. Zhai, R. Xu, J. Dai, X. Ma, X. Yu, Q. Li, Y. Ma, *J. Am. Chem. Soc.* **2022**, *144*, 21640–21647.
- [8] A. F. Young, C. Sanloup, E. Gregoryanz, S. Scandolo, R. J. Hemley, H. Mao, *Phys. Rev. Lett.* **2006**, *96*, 155501.
- [9] M. Bykov, S. Chariton, H. Fei, T. Fedotenko, G. Aprilis, A. V. Ponomareva, F. Tasnádi, I. A. Abrikosov, B. Merle, P. Feldner, S. Vogel, W. Schnick, V. B. Prakapenka, E. Greenberg, M. Hanfland, A. Pakhomova, H.-P. Liermann, T. Katsura, N. Dubrovinskaia, L. Dubrovinsky, *Nat Commun* **2019**, *10*, 2994.
- [10] M. Bykov, E. Bykova, A. V. Ponomareva, I. A. Abrikosov, S. Chariton, V. B. Prakapenka, M. F. Mahmood, L. Dubrovinsky, A. F. Goncharov, *Angewandte Chemie International Edition* **2021**, *60*, 9003–9008.
- [11] D. Laniel, B. Winkler, E. Koemets, T. Fedotenko, M. Bykov, E. Bykova, L. Dubrovinsky, N. Dubrovinskaia, *Nat Commun* **2019**, *10*, 4515.
- [12] M. Bykov, S. Chariton, E. Bykova, S. Khandarkhaeva, T. Fedotenko, A. V. Ponomareva, J. Tidholm, F. Tasnádi, I. A. Abrikosov, P. Sedmak, V. Prakapenka, M. Hanfland, H.-P. Liermann, M. Mahmood, A. F. Goncharov, N. Dubrovinskaia, L. Dubrovinsky, *Angewandte Chemie* **2020**, *132*, 10407–10412.
- [13] A. Aslandukov, F. Trybel, A. Aslandukova, D. Laniel, T. Fedotenko, S. Khandarkhaeva, G. Aprilis, C. Giacobbe, E. Lawrence Bright, I. A. Abrikosov, L. Dubrovinsky, N. Dubrovinskaia, *Angewandte Chemie International Edition* **2022**, *61*, e202207469.
- [14] D. Laniel, F. Trybel, Y. Yin, T. Fedotenko, S. Khandarkhaeva, A. Aslandukov, G. Aprilis, A. I. Abrikosov, T. Bin Masood, C. Giacobbe, E. Bright, K. Glazyrin, M. Hanfland, J. Wright, I. Hotz, I. Abrikosov, L. Dubrovinsky, N. Dubrovinskaia, *Nature Chemistry* **2023**, *1–6*.
- [15] N. P. Salke, K. Xia, S. Fu, Y. Zhang, E. Greenberg, V. B. Prakapenka, J. Liu, J. Sun, J.-F. Lin, *Physical Review Letters* **2021**, *126*, 065702.
- [16] Q. Wei, C. Zhao, M. Zhang, H. Yan, B. Wei, *Physics Letters A* **2019**, *383*, 2429–2435.
- [17] S. Liu, R. Liu, H. Li, Z. Yao, X. Shi, P. Wang, B. Liu, *Inorg. Chem.* **2021**, *60*, 14022–14030.
- [18] J. Jiang, P. Yan, Y. Zhou, Z. Cheng, X. Cui, Y. Ge, Q. Xu, *Advanced Energy Materials* **2020**, *10*, 2002214.
- [19] M. Hasegawa, T. Yagi, *Solid State Communications* **2005**, *135*, 294–297.
- [20] M. B. Lourenço, M. D. Carvalho, P. Fonseca, T. Gasche, G. Evans, M. Godinho, M. M. Cruz, *Journal of Alloys and Compounds* **2014**, *612*, 176–182.
- [21] D. Laniel, A. Dewaele, G. Garbarino, *Inorg. Chem.* **2018**, *57*, 6245–6251.
- [22] A. Salamat, A. L. Hector, P. Kroll, P. F. McMillan, *Coordination Chemistry Reviews* **2013**, *257*, 2063–2072.
- [23] L. Wu, R. Tian, B. Wan, H. Liu, N. Gong, P. Chen, T. Shen, Y. Yao, H.

- Gou, F. Gao, *Chem. Mater.* **2018**, *30*, 8476–8485.
- [24] Deposition numbers  
<https://www.ccdc.cam.ac.uk/services/structures?id=doi:10.1002/chem.202400536> CSD 2325509 (for Pnma-Co3N2), 2325510 (P63/mmc-CoN), 2325511 (for Pnma-CoN), 2308801 (Marcasite-type-CoN2), 2325513 (for Pnma-CoN3), 2325514 (for C2c-CoN5) contain the supplementary crystallographic data for this paper. These data are provided free of charge by the joint Cambridge Crystallographic Data Centre and Fachinformationszentrum Karlsruhe Access Structures service <http://www.ccdc.cam.ac.uk/structures>.
- [25] M. Bykov, E. Bykova, A. V. Ponomareva, F. Tasnádi, S. Chariton, V. B. Prakapenka, K. Glazyrin, J. S. Smith, M. F. Mahmood, I. A. Abrikosov, A. F. Goncharov, *ACS Nano* **2021**, *15*, 13539–13546.
- [26] G. V. Vajenine, G. Auffermann, Y. Prots, W. Schnelle, R. K. Kremer, A. Simon, R. Kniep, *Inorg. Chem.* **2001**, *40*, 4866–4870.
- [27] D. Laniel, B. Winkler, T. Fedotenko, A. Pakhomova, S. Chariton, V. Milman, V. Prakapenka, L. Dubrovinsky, N. Dubrovinskaia, *Phys. Rev. Lett.* **2020**, *124*, 216001.
- [28] M. I. Eremets, A. G. Gavriluk, I. A. Trojan, D. A. Dzivenko, R. Boehler, *Nature Mater* **2004**, *3*, 558–563.
- [29] K. Niwa, D. Dzivenko, K. Suzuki, R. Riedel, I. Troyan, M. Eremets, M. Hasegawa, *Inorg. Chem.* **2014**, *53*, 697–699.
- [30] E. Gregoryanz, A. F. Goncharov, R. J. Hemley, H. Mao, M. Somayazulu, G. Shen, *Phys. Rev. B* **2002**, *66*, 224108.
- [31] C. Ji, A. A. Adeleke, L. Yang, B. Wan, H. Gou, Y. Yao, B. Li, Y. Meng, J. S. Smith, V. B. Prakapenka, *Science advances* **2020**, *6*, eaba9206.
- [32] J. Ibáñez, T. Woźniak, F. Dybala, R. Oliva, S. Hernández, R. Kudrawiec, *Sci Rep* **2018**, *8*, 12757.
- [33] K. Miwa, *Phys. Rev. B* **2018**, *97*, 075143.
- [34] L. Pauling, *J. Am. Chem. Soc.* **1929**, *51*, 1010–1026.
- [35] V. B. Prakapenka, A. Kubo, A. Kuznetsov, A. Laskin, O. Shkurikhin, P. Dera, M. L. Rivers, S. R. Sutton, *High Pressure Research* **2008**, *28*, 225–235.
- [36] M. Mezouar, W. A. Crichton, S. Bauchau, F. Thurel, H. Witsch, F. Torrecillas, G. Blattmann, P. Marion, Y. Dabin, J. Chavanne, O. Hignette, C. Morawe, C. Borel, *J. Synchrotron Rad* **2005**, *12*, 659–664.
- [37] C. Prescher, V. B. Prakapenka, *High Pressure Research* **2015**, *35*, 223–230.
- [38] P. R. O. CrysAlis, *Yarnton, England* **2009**.
- [39] O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. a. K. Howard, H. Puschmann, *J Appl Cryst* **2009**, *42*, 339–341.
- [40] G. M. Sheldrick, *Acta Cryst A* **2008**, *64*, 112–122.
- [41] B. H. Toby, R. B. Von Dreele, *J Appl Cryst* **2013**, *46*, 544–549.
- [42] R. Hrubciak, J. S. Smith, G. Shen, *Review of Scientific Instruments* **2019**, *90*, 025109.
- [43] A. F. Goncharov, N. Holtgrewe, G. Qian, C. Hu, A. R. Oganov, M. Somayazulu, E. Stavrou, C. J. Pickard, A. Berlie, F. Yen, M. Mahmood, S. S. Lobanov, Z. Konôpková, V. B. Prakapenka, *The Journal of Chemical Physics* **2015**, *142*, 214308.
- [44] N. Holtgrewe, E. Greenberg, C. Prescher, V. B. Prakapenka, A. F. Goncharov, *High Pressure Research* **2019**, *39*, 457–470.
- [45] A. Steele, M. D. Fries, H. E. F. Amundsen, B. O. Mysen, M. L. Fogel, M. Schweizer, N. Z. Boctor, *Meteoritics & Planetary Science* **2007**, *42*, 1549–1566.
- [46] P. Hohenberg, W. Kohn, *Phys. Rev.* **1964**, *136*, B864.
- [47] W. Kohn, L. J. Sham, *Phys. Rev.* **1965**, *140*, A1133.
- [48] G. Kresse, J. Furthmüller, *Phys. Rev. B* **1996**, *54*, 11169–11186.
- [49] J. P. Perdew, K. Burke, M. Ernzerhof, *Phys. Rev. Lett.* **1996**, *77*, 3865–3868.
- [50] A. Togo, L. Chaput, T. Tadano, I. Tanaka, *J. Phys.: Condens. Matter* **2023**, *35*, 353001.
- [51] D. Laniel, F. Trybel, A. Aslandukov, J. Spender, U. Ranieri, T. Fedotenko, K. Glazyrin, E. L. Bright, S. Chariton, V. B. Prakapenka, I. A. Abrikosov, L. Dubrovinsky, N. Dubrovinskaia, *Nat Commun* **2023**, *14*, 6207.
- [52] A. Aslandukov, M. Aslandukov, N. Dubrovinskaia, L. Dubrovinsky, *J Appl Cryst* **2022**, *55*, 1383–1391.
- [53] D. Antonangeli, L. R. Benedetti, D. L. Farber, G. Steinle-Neumann, A. Auzende, J. Badro, M. Hanfland, M. Krisch, *Applied Physics Letters* **2008**, *92*, 111911.

## Entry for the Table of Contents



The nitridation behaviour of cobalt nitride unfolds along a progressive path under high-pressure conditions, as investigated using a laser-heated diamond anvil cell, combined with synchrotron XRD, Raman spectroscopy, and density functional theory. Our observations reveal a sequence of  $\text{CoN}_6$  octahedral framework transformations, including face-sharing (in  $\text{CoN}$ ), edge-sharing (in  $\text{CoN}_2$  and  $\text{CoN}_3$ ), and isolated (in  $\text{CoN}_5$ ) polyhedra. These changes align with the polymerization of nitrogen within the crystal.