

Design of Deep-UV Nanowire LED with Al₂O₃ Quantum Dots and Step-Graded n-type AlInGa_N Electron Blocking Layer for High Quantum Efficiency

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Abstract— In this paper, we have proposed a novel deep ultra-violet (DUV) AlGa_N/Ga_N nanowire light-emitting diode (LED) with a step-graded n-type AlInGa_N electron blocking layer (EBL) instead of a conventional p-type AlGa_N EBL. The nanowire is designed with Al₂O₃ quantum dots (QDs) deposited all over the substrate. The proposed nanowire is designed for a ~265 nm wavelength emission without affecting the hole injection efficiency. Due to enhanced carrier transport in the step-graded n-type EBL structure, there occurs reduced electron leakage into the p-region, superior hole activation and hole injection, improved output power and internal quantum efficiency (IQE). Moreover, this specially designed EBL reduces the quantum confined stark effect in the active region, ultimately enhancing the carrier wave functions overlap. The device structure is simulated using Atlas technology computer-aided design (TCAD). The efficiency is improved from ~36.48% to ~49.46% while switching from conventional p-type EBL to step-graded n-type EBL. Furthermore, our proposed structure exhibits 1.61% efficiency droop, which is significantly ~4.8 times lower as compared to the regular structure.

Keywords— *Electron blocking layer (EBL); internal quantum efficiency (IQE); light-emitting diode (LED); multi-quantum well (MQW).*

I. INTRODUCTION

DUE to the wide band-gap, aluminium gallium nitride (AlGa_N) based deep ultra-violet (DUV) light-emitting diodes (LED) are capable of replacing the existing incandescent UV light sources for a wide range of applications, such as water purification, pharmaceutical applications, and disinfection [1]–[4]. III-nitrides like gallium nitride (Ga_N), indium nitride (InN), aluminium nitride (AlN) and their alloys have energies ranging from 0.7 eV (InN) through 3.4 eV (Ga_N) to 6.2 eV (AlN), making them uniquely suitable for visible and ultra-violet LEDs. Across the entire UV-A, UV-B, and UV-C spectral range, UV LEDs offer significantly decreased power consumption, low cost, and reduced space [5]–[8].

Optical power and internal quantum efficiency (IQE) are considered to be the critical parameters, which need further improvement so that high-power LEDs can penetrate into the

consumer industry of lighting [9][10]. However it gets hindered by the electron overflow into the p-Ga_N region and the inadequate hole supply into the active region [11][12]. Due to overflow of electrons, the compactness of carriers inside the wells will be minimised which will lead to poor radiative recombination. Therefore to restrict this overflow, a p-type electron blocking layer (EBL) is inserted between the active region and p-Ga_N layers. The electrons are thus confined within the multi-quantum well (MQW) because of the huge potential barrier created by AlGa_N layer, resulting in a low electron overflow. Because of the large defect density and dislocation in the active region, it is still difficult to obtain efficient p-type conduction in large Al content EBL [13]–[16]. This occurs due to various reasons related with magnesium (Mg)-dopants including tremendously large activation energy (till ~600 meV), little tendency to get soluble and self-compensation while forming point defects [17][18]. Because of in-effective Mg-doping, a very poor hole concentration occurs in this Al rich EBL [19]–[21]. Also when the potential barrier height difference is large, it obstructs the hole transportation into the MQWs leading to insufficient hole supply taking part in the radiative recombination with electrons [22]. Therefore in an attempt to minimize electron overflow without compromising the hole injection, a novel LED nanowire structure has been proposed in this paper using step-graded n-type AlInGa_N EBL inserted between the active region and n-AlGa_N cladding layer. This EBL consists of silicon (Si)-doped AlInGa_N permitting effective cooling of hot electrons before they inject into the active region and thus there is a less overflow of electrons. The proposed nanowire LED is surrounded by Al₂O₃ nanoparticles with very low radius which results in low self-heating of the device. In our proposed device, due to step-graded EBL structure, the kinetic energy and electron velocity inserting the active region decrease which in turn reduce the electron mean free path and improve the electron confinement in the EBL [23]. This paper is structured as follows: Section II is dedicated to a brief description of the simulation methodology, presentation of the device structure and parameters. Section III is focused on the presentation

and discussion of the obtained results, and, finally, the conclusions are given in the last section.

II. DEVICE STRUCTURE AND PARAMETERS

Firstly, in order to validate our device model, we have examined the conventional AlGaIn deep UV LED formed on a c-plane AlN template with ~ 284 nm emission wavelength as our reference device and denoted as LED₁. This work was studied and experimentally presented by Yan et al [24]. LED₁ comprises of 3 μm n-Al_{0.6}Ga_{0.4}N layer (Si doping: $5 \times 10^{18} \text{ cm}^{-3}$), followed by an active region, p-Al_{0.65}Ga_{0.35}N EBL (20 nm, Mg doping: $2 \times 10^{19} \text{ cm}^{-3}$) and then capped by a 50 nm p-Al_{0.5}Ga_{0.5}N hole injection layer (Mg doping: $2 \times 10^{19} \text{ cm}^{-3}$), and on the top a 80 nm p-GaN contact layer (Mg doping: $1 \times 10^{20} \text{ cm}^{-3}$). The active region comprises of five

intrinsic Al_{0.4}Ga_{0.6}N quantum wells (QWs of 3 nm) stuffed between six intrinsic Al_{0.5}Ga_{0.5}N quantum barriers (QBs of 12 nm). As illustrated in Fig. 1, the proposed device (LED₂) is designed as a nanowire and has the similar layers as LED₁ except that the uniform p-type AlGaIn EBL is now replaced by step-graded n-type AlInGaIn EBL prior to the active region. Also LED₂ nanowire is designed with Al₂O₃ quantum dots (QDs) deposited all over the substrate. In this nanowire sapphire is used as the substrate due to the benefit of high heat conductivity and allowing for vertical conduction geometry. During the simulation the radius of the nanoparticles is taken as 10 nm. The LED chip has an overall area of $300 \mu\text{m} \times 300 \mu\text{m}$ which is operated under a temperature of 300 K and the internal absorption loss is assumed as 2000 m^{-1} .

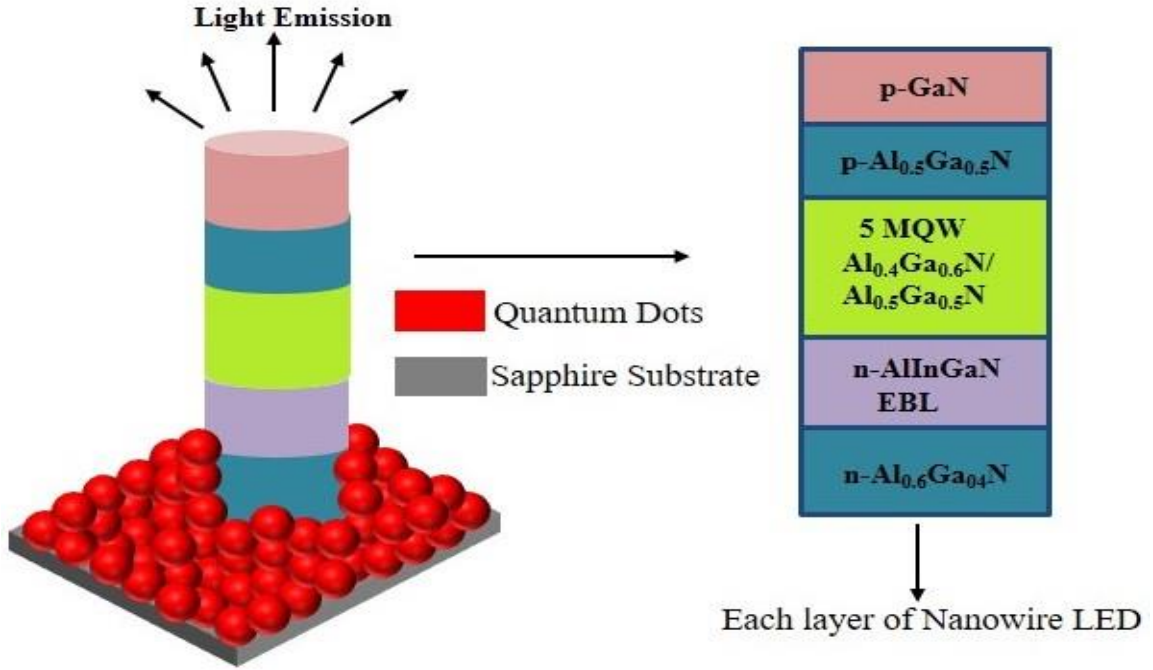


Fig. 1. Schematic diagram of the proposed nanowire LED (LED₂) with n-type EBL and Al₂O₃ nanoparticles deposited over the substrate.

The benefit of n-type EBL consisting of AlInGaIn step-graded layers is the effective cooling of hot electrons before injecting in the active region and thus minimizing overflow of electrons. Also due to the absence of any potential barrier, effective injection of holes can be achieved into the active region. Because of minimized reduced barrier to hole injection, a lower turn-on voltage for n-type EBL is obtained. The EBL of LED₂ is divided into five layers of Al_xIn_yGa_zN; each has a thickness of 4 nm. This layer is uniformly step-graded in such a way that the constituent values of x, y, z is (0.8, 0.03, 0.17), followed by step-wise reduction of Al composition by 10% and 10% increase of gallium composition in each layer (0.7, 0.03, 0.27 for next layer) and so on up to five consecutive divisions till (0.4, 0.03, 0.57) in the last layer.

The band-gap of GaN and AlN are taken as 3.42 eV and 6.2 eV, respectively [25]. The Auger recombination, radiative recombination, SRH recombination coefficient, and light extraction efficiency are taken as $2.88 \times 10^{-30} \text{ cm}^6/\text{s}$, $2.13 \times 10^{-11} \text{ cm}^3/\text{s}$, $6.67 \times 10^7/\text{s}$, and 15%, respectively [26]. Also, the Mg activation energy is scaled linearly from 170 meV to 510 meV for p-Al_xGa_{1-x}N alloy. The electron and hole mobilities are estimated by Cauchy-Thomas

approximation and band energy diagrams are measured by using 6×6 k.p model [26].

III. RESULTS AND DISCUSSION

The device model and parameters which are numerically implemented in our work are optimized using the experimentally obtained data of reference structure i.e., LED₁.

To investigate the performance of the proposed structure, the electroluminescence (EL) spectra of the designed LEDs are displayed in Fig. 2 at an injection current of 40 mA. The light intensity of LED₂ (peak wavelength 260.9 nm) is stronger as compared to LED₁ (peak wavelength 262.8 nm). The presence of a shoulder at ~ 264.5 nm other than the main peak at ~ 262.8 nm for LED₁ is due to significant electron overflow and the resulting parasitic emission from p-AlGaIn layer. Due to decrease in peak wavelength of LED₂, a very minute blue-shift is observed which is apparently assigned to the minimization of quantum-confined stark effect (QCSE) in our final structure. From Fig 2(b), LED₂ shows a prominent blue-shift when compared with the conventional LED₁. The blue shift due to the band-filling effect and screening effect causes the wavelength to initially shift

towards a lower value as the current increases. But when the current continues to rise, the wavelength moves to a longer value known as red shift. This is a result of the thermal effect created by parasitic resistances at the wavelength point where there is a change from blue shift to red shift, as seen by the dashed line in Fig 2(b). In order to understand the underlying mechanism, the energy band diagram and the quasi-Fermi level of the simulated structures are analysed at 40 mA current in Figs. 3(a) and 3(b). The effective conduction band barrier height CBBH (Φ_C) at each barrier

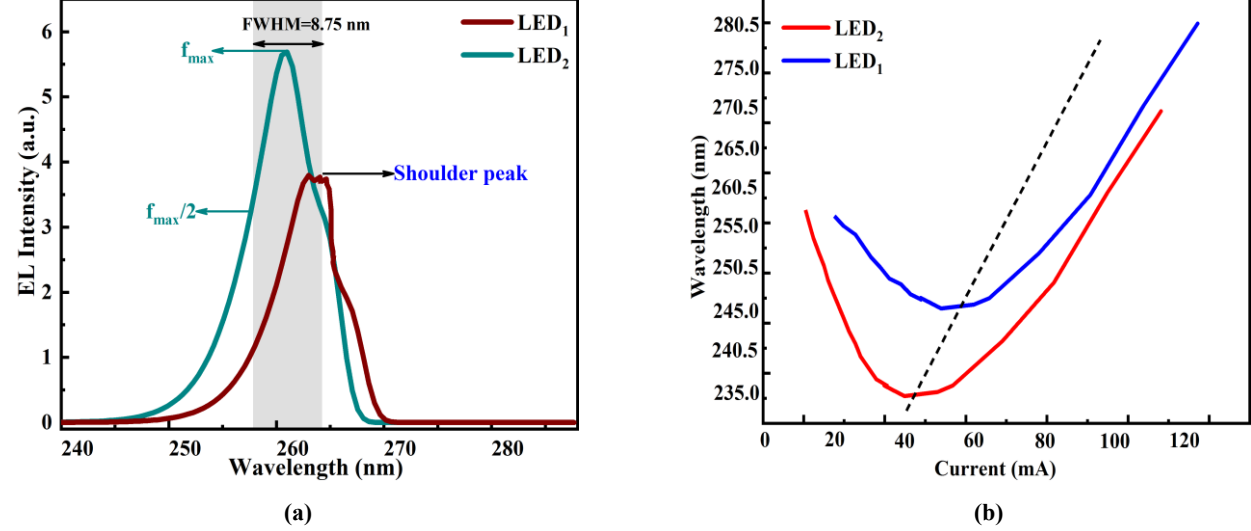


Fig. 2. (a) Electroluminescence (EL) intensity of LED₁ and LED₂ at an injection current of 40 mA, (b) Emission wavelengths as a function of injection current for LED₁ and LED₂.

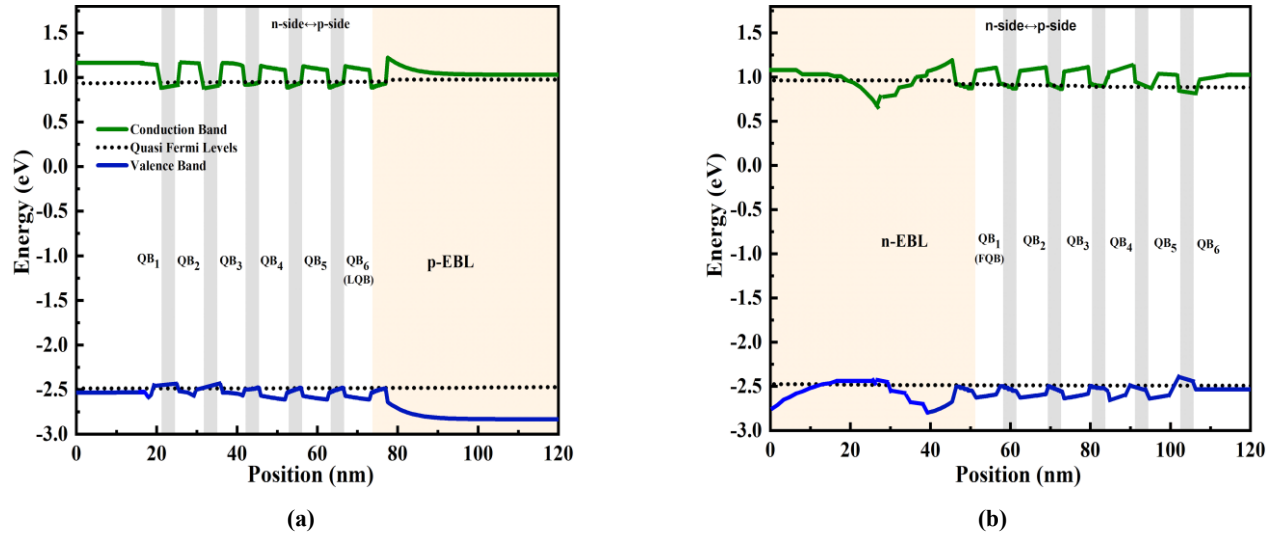


Fig. 3. Energy band profile of (a) the conventional LED (LED₁) and of (b) the proposed LED (LED₂) at 40 mA.

n-EBL presents no blocking effect on the hole transport path into the QWs due to the absence of energy barrier. This leads to the reduction of non-radiative recombination in p-region and improves hole injection into the active region.

TABLE I. EFFECTIVE CBBH (Φ_{EN}), VBBH (Φ_{HN}) OF QBS AND EBL (Φ_{EBL}) FOR LED₁ AND LED₂ (IN MEV)

CBBH	LED ₁	LED ₂	VBBH	LED ₁	LED ₂
Φ_{e2}	107.8	197.1	Φ_{h2}	72.9	75.9
Φ_{e3}	159.1	203.7	Φ_{h3}	76.8	80.8
Φ_{e4}	180.4	312.7	Φ_{h4}	72.2	84.6

(n) and EBL (Φ_{EBL}) of LED₂ is comparatively higher than LED₁ as listed in Table 1. The higher and progressively improved Φ_C in LED₂ constructively confines the electrons in the active region and resists the electron overflow into the p-region. From Fig. 3(a) the p-type EBL generates an energy barrier height of 169.4 meV for holes. On the contrary, in Fig. 3(b), the n-type AlInGa_N builds up an energy barrier of 90 meV for electrons before they enter the MQWs active region. Meanwhile, the

Φ_{e5}	176.6	367	Φ_{h5}	81.6	86.2
Φ_{e6}	174.7	343.9	Φ_{h6}	80.3	107.1
Φ_{EBL}	255.46	368.04			

In LED₂ electrons need to overcome a higher CBBH to reach the p-layer. Because of this, non-radiative recombination of leaked electrons within the holes in the p-side is lowered and availability of holes to be injected towards the active region is more. Consequently a better radiative recombination rate is observed in LED₂, due to stronger carrier concentration, as displayed in Fig. 4. For clear visualization, the position of values in the x-axis is slightly shifted for the two samples. The electron-hole pair

recombination for LED₂ is enormously improved by 93.3% than LED₁ (for reference, third QW is considered).

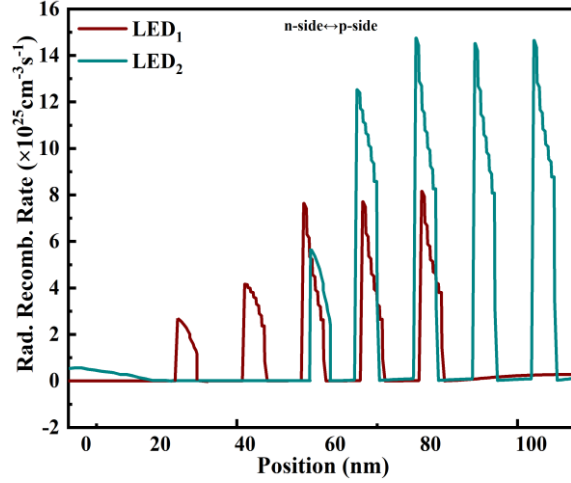


Fig. 4. Radiative recombination rate of both LED₁ and LED₂ at 40 mA.

To understand the carrier transport improvement in the proposed structure (LED₂) we have studied their corresponding IQE and output power performance, as presented in Fig. 5 and 6 respectively. It is apparent that IQE is significantly higher for LED₂ with step-graded n-type EBL by 35.58% at a current injection of 40 mA. This indicates the detrimental impact of p-type EBL on hole injection. Interestingly, the efficiency droop defined as $(\eta_{peak} - \eta_{40\text{ mA}}) / \eta_{peak}$, is severely decreased up to $\sim 1.61\%$ in LED₂, while LED₁ exhibits $\sim 9.42\%$ droop at 40 mA current. This is due to the enhanced carrier transportation and confinement of carriers in the active region of LED₂.

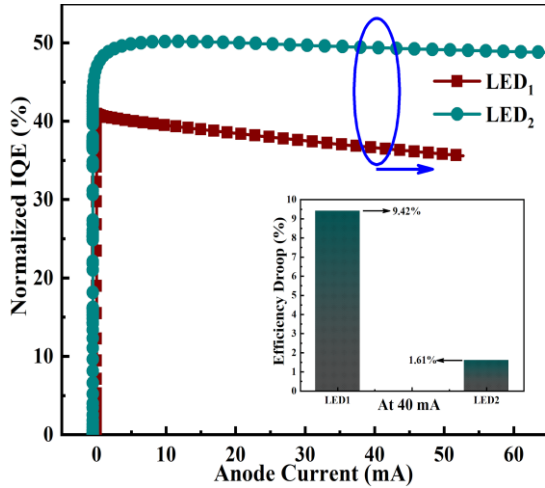


Fig. 5. IQE as a function of injected current in both LED₁ and LED₂. Inset: Efficiency droop for each LED at 40 mA.

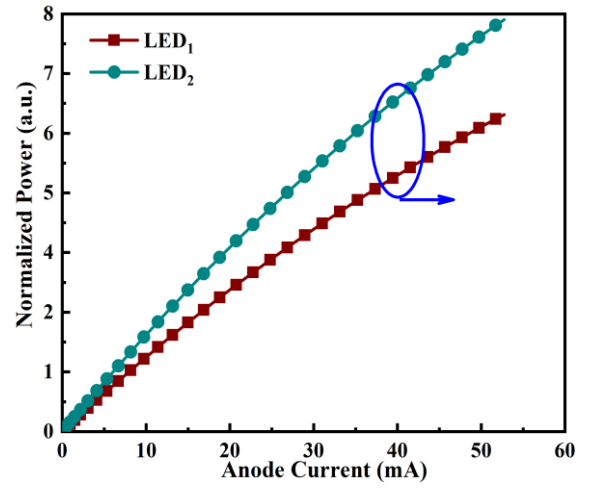


Fig. 6. Output power of both LED₁ and LED₂ as a function of injection current.

The amelioration in the IQE and decrease in droop in LED₂ are attributed to the improved radiative recombination and reduced carrier leakage, owing to superior carrier confinement. This is provided by the step-graded n-type EBL. As shown in Fig. 7, the turn-on voltage of the two structures is nearly the same. However, it is inevitable that LED₂ exhibits a slightly higher operating bias voltage at 40 mA than LED₁ due to the presence of step graded n-type AlInGa_{0.5}N EBL.

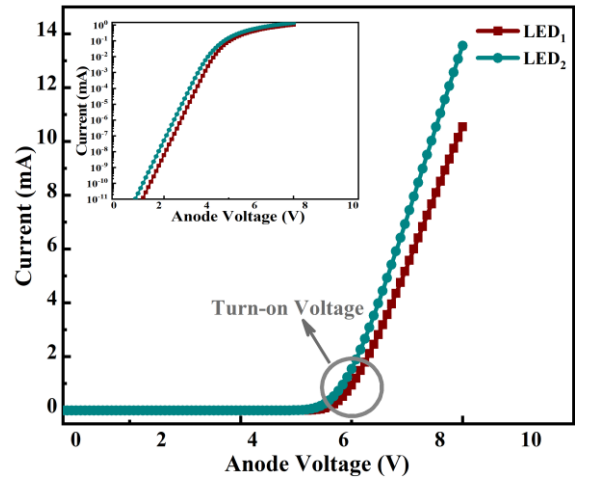


Fig. 7. I-V behavior of both LED₁ and LED₂ in linear scales; Inset: Replotting of the I-V curves in a semi-log scale.

IV. CONCLUSION

In conclusion, we have explored the impact of EBL on the output performance of AlGa_{0.5}N UV LEDs. From the achieved simulation results, it is evident that step-graded n-type AlInGa_{0.5}N EBL has better advantages compared to conventional p-type AlGa_{0.5}N EBL. LED₂ is superior to LED₁ as it reduces electron overflow without compromising hole transport and hole injection into the active region, thereby having a better balance between hole and electron injection in the active region. Step-graded promotes the hole injection by forming negative sheet polarization charges and improves the spatial overlap of the electron-hole wave function. A significant improvement in both IQE (by 23.51%) and power (by 71.48%) of LED₂ is measured compared to LED₁. Furthermore LED₂ is observed to have a minimal efficiency droop of 1.172% and lower electrostatic field. In the future,

extended device performance is expected by optimizing the p-type doping which will further improve the light extraction efficiency along with minimized efficiency droop. While this work is preliminary, continued refinement of this technology platform offers exciting new approaches to improved performance and manufacturing of solid-state lighting systems.

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