

Fabrication of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ -based electrocatalyst for efficient bifunctional electrocatalytic water splitting



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ABSTRACT

Exploring the cost-effective and earth-abundant electrocatalyst for efficient bifunctional activity is very challenging these days. The main reason is very slow OER (oxygen evolution reaction) kinetics as compared with HER (hydrogen evolution reaction). An electrocatalyst with enhanced kinetics towards OER and HER is highly required, which may be attributed to a larger surface area, an increased number of active sites, and lower charge transfer resistance. In this paper, we have reported a very efficient bifunctional $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalyst by a facile ultrasonication route followed by annealing for OER/HER performance in an alkaline solution. The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalyst exhibits a lower overpotential of 228 mV and 270 mV during OER to accomplish a current density of 40 and 100 mAcm^{-2} while it requires a 368 mV and 418 mV overpotential during HER to achieve a current density of 40 and 100 mAcm^{-2} . The prepared $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalyst has a lesser Tafel slope (132 mVdec^{-1} ; OER, 115 mVdec^{-1} ; HER) as compared to $\text{Ni}_2\text{O}_2(\text{OH})$ (239 mVdec^{-1} ; OER, 124 mVdec^{-1} , HER). $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ exhibit reduced charge resistance, enhanced electrochemical active surface area, greater carrier concentration, and a higher exchange current density, which confirms its superior bifunctional electrocatalytic activity. This work provides an economical and affordable means to prepare superior, efficient bifunctional electrocatalysts for widespread commercial applications.

1. Introduction

There are many common issues all around the world. Among these issues, the few issues are very important. These are energy production and storage and environmental pollution. Researchers are continuously working their efforts to address these issues. The use of fossil fuels for energy production and preservation is one solution to energy shortages that has gained widespread recognition [1–3]. On the other hand, when fossil fuels are burned, they give off harmful gases that contribute directly to pollution and, in the long run, global warming [4,5]. Because of this, it is crucial to phase out fossil fuels in favor of renewable energy sources that are better for the environment [6,7]. Hydrogen is a great

choice for making energy in the future because it is light and has a high energy density [8,9]. Hydrogen can be made in several different ways, such as by electrolyzing or photolyzing water, reforming biomass, or burning fossil fuels [10]. Most hydrogen is made through a process called electrolysis [11]. In electrochemical water-splitting reactions, at cathode, the HER takes while the OER takes place at anode. [12–14]. Rare and expensive metals are needed for HER and OER technologies, specifically platinum (Pt) for the electrochemical HER process and iridium and ruthenium dioxide (IrO_2 ; RuO_2) for the OER process [15–17]. Electrolysis needs more energy because the reactions at the electrode junctions take place slowly. This is because the overpotential is high. To get around the problems with traditional noble metal

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electrodes, we need to find a catalyst that is effective, earth-abundant, cheap, and long-lasting [18,19]. Numerous studies have demonstrated the potential of metal nitrides, metal oxides, metal hydroxides, metal oxide hydroxides, metal layer double hydroxides, metal sulfides, and transition metal dichalcogenides as electrocatalysts, but none of them have been able to match the stability and long-term catalytic efficiency of catalysts made from noble metals [20–23]. In the same electrochemical device, the base electrode must therefore effectively serve as a dual-functional catalyst material for both HER and OER processes [24]. Each electrocatalyst requires a different preparation process, which raises the price of both the engineering and manufacturing phases. An electrode material with a large surface area and built-in catalytic activity would be ideal for splitting water [25]. It is well known that transition metals like Au, Ni, Co, Fe, and Cu have the best electrocatalytic properties after the platinum group metals [26]. Ni-based alloys and nanocomposite materials have drawn more consideration because of their potential in both alkaline and acidic solutions. The overpotential of Ni-based composites is low, and they are also resistant to corrosion [27,28].

The nickel oxide hydroxide ($\text{Ni}_2\text{O}_2(\text{OH})$) as an electrocatalyst for water splitting has also received significant attention due to its material potential. There is a lot of surface area available for electrocatalytic reactions on $\text{Ni}_2\text{O}_2(\text{OH})$ because of its layered structure. Their electrocatalytic activity and efficiency for water-splitting reactions, especially the OER, are improved as a result of layered structure. The OER reaction is an essential part of the water-splitting reaction, but it is a difficult reaction that calls for a lot of overpotential to be active. $\text{Ni}_2\text{O}_2(\text{OH})$ are easy to synthesize, abundant, and inexpensive, making them preferable over electrocatalysts based on precious metals like platinum for use in large-scale applications. $\text{Ni}_2\text{O}_2(\text{OH})$ activity for the HER is comparable to that of $\text{Ni}(\text{OH})_2$ and $\text{NiO}(\text{OH})$, but lower than that of some other electrocatalysts [29,30]. Researchers have attempted a variety of methods, including doping with other metals and integrating them into composite materials with other materials like carbon nanotubes (CNTs), to increase their activity for HER [31].

CNTs are appealing candidates for use as a support material for electrocatalysts due to their high electrical conductivity, massive surface area, and unique structural properties [2,32]. Composites for electrochemical water splitting that contain CNTs offer multiple distinct advantages. To begin with, CNTs have high electrical conductivity, which helps with charge transfer in the water-splitting process. This increases the entire process's efficiency by contributing to electron transfer. Secondly, the incorporation of CNTs into composites improves the stability and longevity of the electrode material. Also, it ensures that it will work well for a long time. The unique structure of CNTs provides a large surface area, which makes active water-splitting sites more available and thus increases efficiency. In the last analysis, CNTs are good catalyst supports. By helping water-splitting reactions involve active species, this also promotes the catalytic activity of the composite electrode material as a whole. When CNTs are incorporated with $\text{Ni}_2\text{O}_2(\text{OH})$, their electrocatalytic properties for both OER and HER are enhanced. The resulting composite material can show synergistic effects, in which the combined properties of the two materials improve the electrocatalytic activity as a whole. The $\text{Ni}_2\text{O}_2(\text{OH})$ nanoparticles can be stabilized by the CNTs, increasing their longevity and stability during electrocatalytic reactions, and the CNTs might offer a conductive pathway for electron transfer [33].

$\text{Ni}_2\text{O}_2(\text{OH})$ can be made using a variety of techniques. In one of the usual methods used, when nickel hydroxide is precipitated from aqueous nickel salt solutions in another preparation, thermal treatment leads to the precipitation of $\text{Ni}_2\text{O}_2(\text{OH})$. Another method is hydrothermal synthesis which involves placing nickel precursors in an aqueous solution and sealing them at elevated temperatures and pressures to promote the synthesis of $\text{Ni}_2\text{O}_2(\text{OH})$. Moreover, deposition by electrochemical techniques includes first electrodeposited and then oxidized nickel on a conductive substrate so it turns into $\text{Ni}_2\text{O}_2(\text{OH})$. Finally, the preparation

of $\text{Ni}_2\text{O}_2(\text{OH})$ can also be controlled by using sol-gel method which requires a sol containing nickel ions and gelling it, thereafter thermally treating this varied range of synthesis procedures makes it feasible to tune the properties of nickel hydroxide for usage as a catalyst in industry, energy storage, and electrochemical sensors. There are benefits and drawbacks associated with each of them. Ultrasonication stands out from these methods because the chemical effects result from acoustic waves. The interfacial implosion of bubbles in the solution causes a sudden pressure signal and a high temperature in the solution. This acoustic cavitation resulted in localized, extreme micro-mixing [34]. The rate of the ultrasonic chemical reaction is so high because of the numerous times the reactant molecules collide, which is one of the benefits of sonochemical synthesis. This method does not require the use of any additional external reagents, such as capping or surfactant reagents. Additionally, it is because the products of sonochemical synthesis are highly pure and the reaction environment is more environmentally friendly. Furthermore, it is simple to control the size distribution of products by simply changing the ultrasound frequency and power [35–38].

In this study, $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalysts were fabricated by ultrasonication, followed by annealing of the materials. The resultant $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ exhibits excellent catalytic performance for electrochemical water-splitting reactions (HER/OER) in an alkaline medium. The notable electrocatalytic performance of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst can be primarily attributed to its distinctive porous structure, excellent conductivity, and expansive surface area, which are derived from the presence of carbon nanotubes.

2. Experimental Section

2.1. $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ Synthesis

A quick, low-cost, and easy technique of ultrasonic irradiation was used to prepare the $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalysts. 6.87 mmol $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and 24 g/20 mL polyethylene glycol (PEG) were co-dissolved in a crucible. Two hours of ultrasonic treatment was applied to the mixed solution. Afterward, the ultrasonically treated mixed solution was annealed at 700 °C for 2 h. The resultant $\text{Ni}_2\text{O}_2(\text{OH})$ material was acquired after annealing. The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ nano-hybrid was then synthesized by using a standard ultrasonication method. The as-prepared $\text{Ni}_2\text{O}_2(\text{OH})$ and CNTs suspension (1mg/1 mL) were co-dissolved (9:1) and placed in an ultrasonic cleaner bath for 120 min of ultrasonic processing, thereafter obtained mixture was dried at 80 °C and then used for further electrocatalytic applications. The overall synthesis mechanism and its electrochemical characterization are pictorially represented in Fig. 1.

2.2. Materials characterization

$\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ samples were characterized by Lab XRD-6100 powder X-ray diffraction (XRD) to determine their crystal phase, structure, and crystallite size. In the 20–60° wavelength range, $\text{K}\alpha$ radiations from a Cu source with a wavelength of 1.54 Å were utilized in the analysis. The molecular composition and chemical bonding of the electrocatalysts were monitored using Fourier transform infrared spectroscopy (FTIR; IR affinity 1S). The fabricated samples were analyzed by employing a scanning electron microscope (SEM; Inspect S50 FEI) to determine their structure and surface morphology. The XPS measurements were performed using a Kratos Amicus/ESCA 3400 instrument. The samples were irradiated with 240 W unmonochromated Mg $\text{K}\alpha$ X-rays, and photoelectrons emitted at 0° from the surface normal were energy analyzed using a DuPont type analyzer. The pass energy was set at 150 eV. CasaXPS was used to process raw data files.

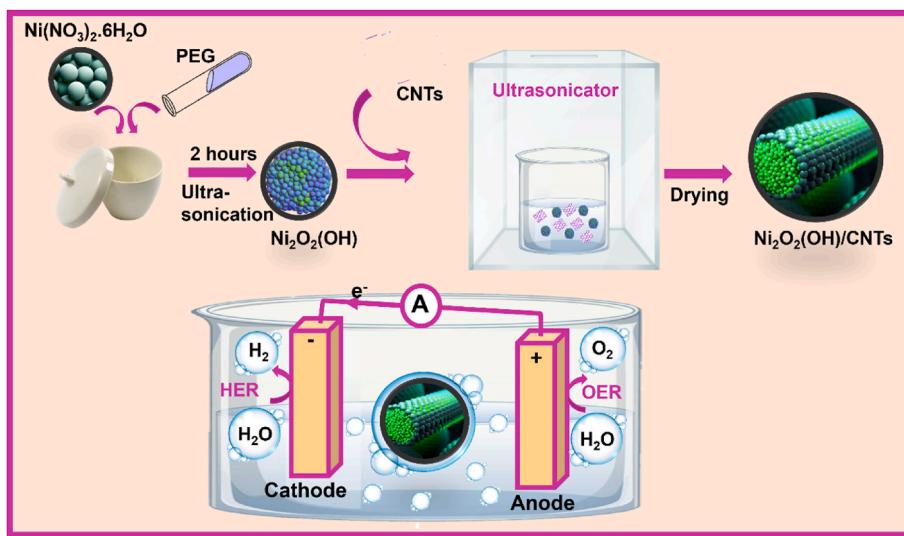


Fig. 1. Synthesis and mechanism scheme of low-cost bifunctional $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalyst for OER/HER electrocatalysis.

2.3. Preparation of electrodes

The $1\text{ cm} \times 1\text{ cm}$ measured substrate material (carbon fiber cloth, CFC) was washed firstly with ethanol, acetone, and deionized water. The catalyst ink was prepared by sonicating the 15 mg of catalyst into $100\text{ }\mu\text{L}$ of deionized water for 2 h. The prepared catalyst ink was then mounted on a premeasured CFC selective area by drop casting method.

2.4. Electrochemical assessments

The electrocatalytic activities of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalysts were studied by using a Gamry Interface 5000E electrochemical workstation. For electrochemical analysis, three-electrode configurations were established: a reference (mercury/mercury oxide; Hg/HgO), a counter (platinum foil; Pt), and a working (modified and unmodified CFC) electrode. The electrolyte solution of 1 M KOH was used in the electrochemical tests. The OER activity of $\text{Ni}_2\text{O}_2(\text{OH})$, $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$, and RuO_2 electrocatalysts were evaluated by using linear sweep voltammetry (LSV) polarization curves at a sweeping rate of 5 mVs^{-1} . To better understand the HER process, LSV recording of as-synthesized materials were made at 5 mVs^{-1} and compared to LSV recordings of CFC, CFC with binder, and Pt foil. In light of the LSV curve, we can determine the Tafel plot ($\log j$ vs overpotential) by using the equation; $\eta = b \log j + a$, where η represents overpotential, j denotes current density, b refers to tafel slope and a is the intercept [39]. The obtained $E_{\text{Hg}/\text{HgO}}$ was converted into reversible hydrogen electrode potential by using this equation: $E_{\text{RHE}} = E_{\text{Hg}/\text{HgO}} + 0.098 + 0.059\text{ pH}$ and the corresponding overpotential (OP) in terms of OER was estimated by subtracting the thermodynamic potential value, which is 1.23 V from the reference electrode potential ($E_{\text{Hg}/\text{HgO}}$) [40]. The electrochemical active surface area (ECSA) of resultant samples was quantified using cyclic voltammograms (CV) at various scan rates, with the most common being the non-Faradic current region at CV fixed potential range. The ECSA of both $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ materials are calculated by dividing the C_{dl} value of each sample by a specific capacitance value (0.040 mFcm^{-2}) [41]. Nyquist plots from electrochemical impedance spectroscopy (EIS) tests were considered to further evaluate the catalytic efficiency of as-fabricated materials. Chrono-potentiometric analysis was performed to assess the stability of the electrocatalyst for OER electrocatalysis, while chronoamperometric analysis was considered to check the stability of the material during HER.

3. Results and discussion

3.1. XRD studies

The XRD technique was performed to analyze the composition, crystal system, and crystal structure of the prepared samples. The diffraction patterns of prepared $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ nanostructures are displayed in Fig. 2. The diffraction peaks at $2\theta = 30.2^\circ$, 31.46° , 35.18° , 40.38° , 44.26° , 45.27° , 50° , 53.6° , 56.28° , and 58.46° are well-indexed with crystal planes (104), (011), (200), (203), (114), (015), (212), (116), (302), and (215) respectively. The diffraction patterns for $\text{Ni}_2\text{O}_2(\text{OH})$ correspond to an orthorhombic crystal system with space group Pnnm (JCPDS# 01-084-1459). The peak at 29° in the XRD of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ nanocomposites corresponds to the (102) diffraction plane of water molecules, as indicated by standard JCPDS#00-016-0598. A clear diffraction spike at 25.7° is observed, which is the characteristic peak of CNTs [42] in the XRD patterns of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ which confirms the successful formation of a binary nanocomposite.

The X-ray density of as-prepared $\text{Ni}_2\text{O}_2(\text{OH})$ was calculated by employing the following equation [43].

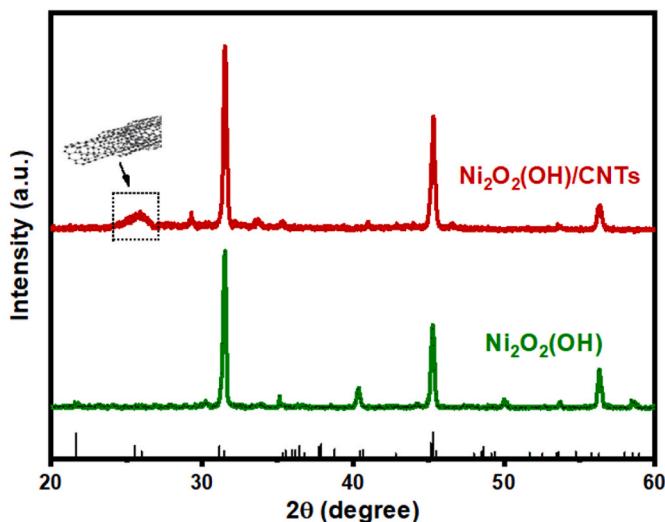


Fig. 2. XRD patterns of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$.

$$d_x = \frac{Z \times M}{N_A \times V} \quad (1)$$

Whereas d_x denotes the X-ray density of a given material, M is the molar mass of the material, Z represents the total number of molecules present in a unit cell volume, V shows the total volume of a unit cell, and N_A denotes Avogadro's number. The calculated d_x of $\text{Ni}_2\text{O}_2(\text{OH})$ is 5.409 g cm^{-3} .

The specific surface area (SSA) of $\text{Ni}_2\text{O}_2(\text{OH})$ was determined by using the following equation [44]:

$$\text{SSA} = A_p/d_x \times V_p \quad (2)$$

In this equation, the specific area of the particle is denoted by A_p , the volume of the particle is represented by V_p and d_x refers to the X-ray density of material. The fabricated $\text{Ni}_2\text{O}_2(\text{OH})$ has a greater specific surface area ($399.68 \text{ m}^2 \text{ g}^{-1}$), which is calculated by using the above equation. The higher SSA of $\text{Ni}_2\text{O}_2(\text{OH})$ is due to its sonochemical synthesis, as the acoustic cavitation generated during sonochemical synthesis promotes the formation of small and uniform nanoparticles with a high surface area-to-volume ratio.

The crystallite size of the prepared $\text{Ni}_2\text{O}_2(\text{OH})$ was also determined by using the Debye-Scherrer formula [45]. The resultant $\text{Ni}_2\text{O}_2(\text{OH})$ estimated crystallite size is given in Table 1.

The dislocation density of the resultant $\text{Ni}_2\text{O}_2(\text{OH})$ was computed by using the following relation.

$$\delta = 1/D^2 \quad (3)$$

Where δ refers to dislocation density and D stands for crystallite size. The dislocation density of the $\text{Ni}_2\text{O}_2(\text{OH})$ sample is 0.000826 nm^{-2} .

3.2. FTIR analysis

FTIR study was done to examine the molecular nature and composition of the as-prepared $\text{Ni}_2\text{O}_2(\text{OH})$ material. All FTIR spectra were recorded at room temperature. The FTIR study was carried out in wavenumber range from about 500 to 4000 cm^{-1} . FTIR spectra of prepared compound $\text{Ni}_2\text{O}_2(\text{OH})$ and its composite with carbon nanotubes (CNTs) are shown in Fig. 3. The absorption band located at 651 cm^{-1} corresponds to the metal-oxygen vibrational mode [46]. The appearance of bands between 800 and 1800 cm^{-1} range could be assigned to anionic impurities that may not have been completely removed during the washing. The bands located at 1095 cm^{-1} , 1199 cm^{-1} , and 1459 cm^{-1} are attributed to carbonate group stretching vibrations, which may appear from the adsorption of atmospheric CO_2 onto the material [46,47]. Detected band around 1694 cm^{-1} was ascribed to the bending vibrations of water molecules [48]. Two bands observed in the FTIR spectrum at 3460 cm^{-1} and 3734 cm^{-1} could be allocated to different stretching modes of OH [46]. FTIR spectra of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ reveals that composite material has nearly the same features as the bare $\text{Ni}_2\text{O}_2(\text{OH})$ material with variations in band intensity and positions. These changes indicate the formation of the composite material, signifying interactions between the $\text{Ni}_2\text{O}_2(\text{OH})$ and CNTs components.

Table 1
XRD parameters of as-fabricated $\text{Ni}_2\text{O}_2(\text{OH})$.

Sr. No.	Parameters	$\text{Ni}_2\text{O}_2(\text{OH})$
1.	Crystal phase	Orthorhombic
2.	Crystallite size (nm)	34.8
3.	Dislocation density (nm^{-2})	0.000826
4.	Lattice constant $a(\text{\AA})$	5.07
5.	Lattice constant $b(\text{\AA})$	2.9
6.	Lattice constant $c(\text{\AA})$	13.9
7.	Cell volume (\AA^3)	204.37
8.	Specific surface area ($\text{m}^2 \text{ g}^{-1}$)	399.68
9.	X-ray density (g cm^{-3})	5.409

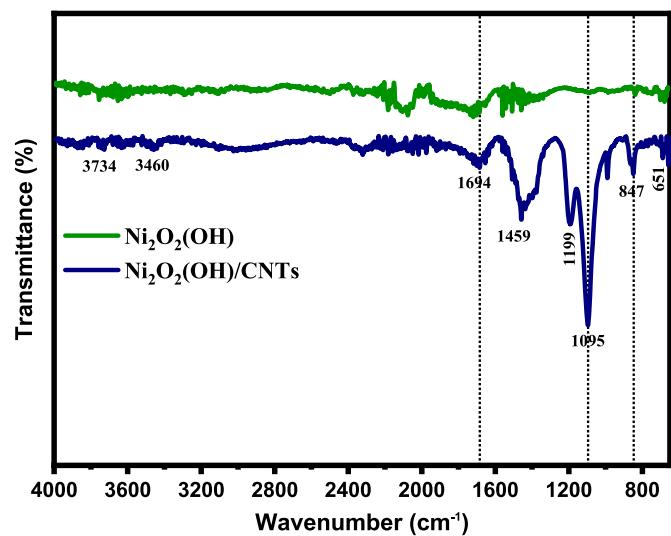


Fig. 3. FTIR spectra of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$.

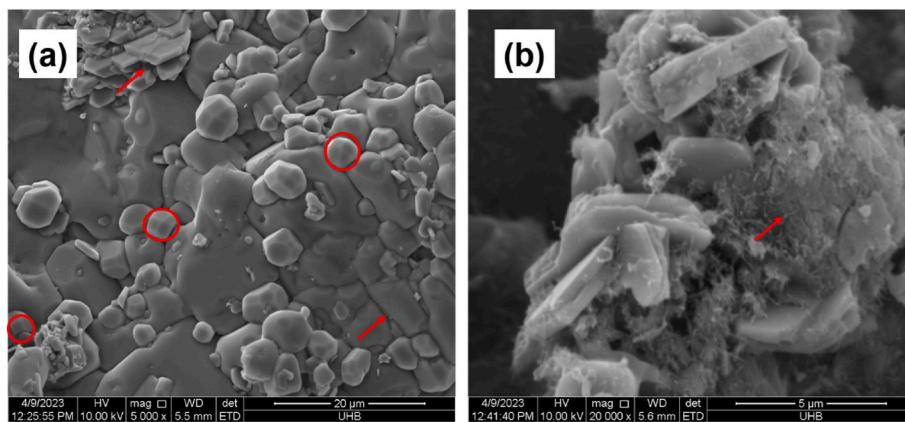
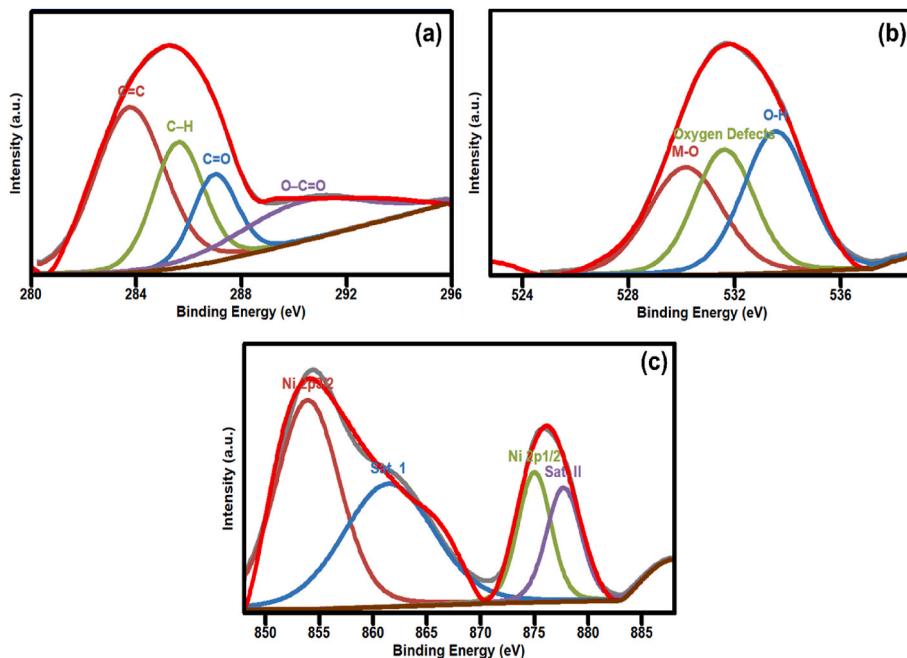
3.3. SEM analysis

The SEM images are recorded through the morphological analysis of as-prepared samples. Fig. 3(a) and (b) represent SEM images of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$, respectively. The SEM images reveal various particles of $\text{Ni}_2\text{O}_2(\text{OH})$ in Fig. 4(a). Some particles appear as elongated nanoplates, exhibiting a thin, plate-like morphology. These nanoplates are elongated along a particular crystallographic direction, resulting in an anisotropic shape. Intermingled with the nanoplates are polyhedral-shaped particles. These particles have well-defined, geometric shapes with multiple facets, resembling polyhedrons. The nanoplates and polyhedral particles are distributed throughout the SEM micrograph, creating a heterogeneous population of $\text{Ni}_2\text{O}_2(\text{OH})$ particles. Fig. 4(b) shows the clear presence of thread-like structures in $\text{Ni}_2\text{O}_2(\text{OH})$, which confirms the formation of CNTs-based composites. In the $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ sample, the lateral sides of plate-like structures are also visible, while polyhedron-like structures are covered by a dense network of CNTs, indicating a tailored architecture promoting enhanced surface area and active sites for electrocatalytic activity.

3.4. X-ray photoelectron spectroscopic (XPS) analysis

Exploring the surface chemical composition of the as-synthesized $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ involved a thorough investigation through X-ray photoelectron spectroscopy (XPS). The high-resolution curve-fitted XPS peaks corresponding to C 1s, O 1s, and Ni 2p are delineated in Fig. 5 (a–c). In Fig. 5(a), the C 1s core level presents a nuanced breakdown with four components identified as C=C at 284.4 eV , C-H at 285.3 eV , C=O at 286.7 eV , and O-C=O at 289.0 eV [49]. Analysis of the O 1s XPS spectra in Fig. 5(b) reveals three distinct bound states (M – O, oxygen defects, and O–H) associated with lattice oxygen, deficient oxygen sites, and C–O bonds within the carbon matrix or arising from the inevitable physical adsorption of water molecules on the surface [50, 51].

The high-resolution XPS spectrum of Ni 2p (Fig. 5(c)) presents a detailed insight into the electronic structure of nickel in the synthesized $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$. Two spin-orbit doublets at binding energy values of 855.8 and 873.4 eV are discerned, featuring a spin energy separation of 17.6 eV , indicative of Ni 2p $3/2$ and Ni 2p $1/2$ components. The peaks at 861.7 and 879.6 eV are identified as the shake-ups of Ni 2p $3/2$ and Ni 2p $1/2$, respectively, providing valuable information on the chemical environment and electronic configuration of nickel within the material [52]. This comprehensive XPS analysis not only defines the elemental composition but also explores the chemical states and bonding

Fig. 4. SEM images of (a) $\text{Ni}_2\text{O}_2(\text{OH})$ and, (b) $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$.Fig. 5. XPS spectra of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ (a) C 1s, (b) O 1s and (c) Ni 2p.

configurations, thereby advancing our understanding of the material's intricate properties.

3.5. OER and HER studies

Cyclic voltammetry profiles were used to figure out the double-layer capacitance (C_{dl}) so that the active sites of electrocatalysts could be evaluated. Fig. 6 represents C_{dl} plots (I vs. scan rate) of as-fabricated materials. The average C_{dl} of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ samples was calculated as 0.165 and 0.24 cm^{-2} , respectively. The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst has a higher value of average C_{dl} than the corresponding $\text{Ni}_2\text{O}_2(\text{OH})$ which demonstrates greater exposure of active sites for OER/HER. The ECSA of as-prepared samples was also determined by using the average C_{dl} values and is presented in Table 3. The ECSA was assumed to be proportionally correlated with the catalytic efficiency. The greater ECSA value of the $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst indicates that it provides enormous surface area and active sites for intrinsic electrocatalytic activities as compared to its counterpart.

LSV was employed with a standard three-electrode configuration in an alkaline medium at a potential window of 1.0–1.8 V (vs. RHE) to measure the electrochemical OER activities of the as-prepared materials

(Fig. 7(a)) and determine the synergistic effects of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ composites. The onset potential of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ (1.46 V vs. RHE) is lower than the pristine $\text{Ni}_2\text{O}_2(\text{OH})$ (1.49 V vs. RHE). The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ requires 228 mV of overpotential to deliver the current density of 40 mA cm^{-2} which is much lesser than $\text{Ni}_2\text{O}_2(\text{OH})$ (249 mV at the current density of 40 mA cm^{-2}) and RuO_2 (270 mV at 40 mA cm^{-2}). The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ requires 270 mV of overpotential to attain the higher current density of 100 mA cm^{-2} which is much lesser than $\text{Ni}_2\text{O}_2(\text{OH})$ which requires 300 mV and RuO_2 needs 310 mV of overpotential at the current density of 100 mA cm^{-2} . While RuO_2 requires more overpotential to reach a higher current density as compared to others. It indicates the higher OER activity of carbon nanotube-based electrocatalysts. To evaluate the kinetics of electrocatalysts, Tafel slopes are calculated from the Tafel plots as presented in Fig. 7(b). The calculated Tafel slopes of $\text{Ni}_2\text{O}_2(\text{OH})$, $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$, and RuO_2 are 239, 132, and 129 mV dec^{-1} , respectively. The lesser Tafel slope and overpotential value of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ than its counterpart demonstrate that it has more efficient charge kinetics, faster electron mobility, and excellent intrinsic OER activity. The long-term stability of the electrocatalyst is also an important parameter to analyze its electrochemical properties. The stability of the OER $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$

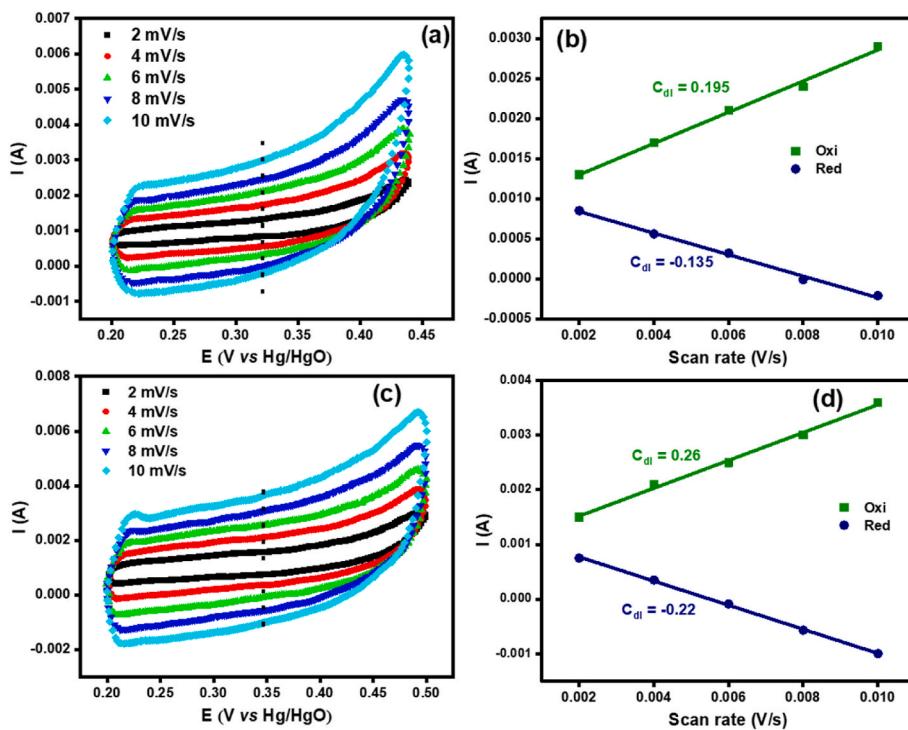


Fig. 6. Cyclic voltammograms of electrocatalysts with varying non-faradaic current regions at various scan rates; (a-b) bare $\text{Ni}_2\text{O}_2(\text{OH})$ and (c-d) $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$.

Table 2

Parameters for the evaluation of bifunctional electrocatalytic activity of prepared materials.

Sr. No.	Materials	Tafel slope (mVdec^{-1})		Overpotential (mV)	
		OER	HER	OER	HER
1.	$\text{Ni}_2\text{O}_2(\text{OH})$	239	124	$\eta_{40} = 249$	$\eta_{40} = 471$
				$\eta_{100} = 300$	$\eta_{100} = 519$
2.	$\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$	132	115	$\eta_{40} = 228$	$\eta_{40} = 368$
				$\eta_{100} = 270$	$\eta_{100} = 418$
3.	RuO_2	129	–	$\eta_{40} = 270$	–
				$\eta_{100} = 310$	–
4.	Pt foil	–	76	–	$\eta_{40} = 157$
				–	$\eta_{100} = 227$

Table 3

Experimental parameters for bifunctional activity evaluation of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalysts.

Sr. No.	Parameters	$\text{Ni}_2\text{O}_2(\text{OH})$	$\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$
1.	Average C_{dl} (cm^{-2})	0.165	0.24
2.	ECFA (cm^2)	4.13	6
3.	R_{ct} (Ω)	5.5	2.9
4.	j^0 (mA cm^{-2})	1.167	2.214
5.	E_{fb} (V)	0.61	0.55
6.	$N_D \times 10^{22}$ (cm^{-3})	7.66	14.7

electrocatalyst was further assessed by chrono-potentiometric measurements for 5 h at a fixed value of current density of 40 mA cm^{-2} as depicted in Fig. 7(d). It indicates that there is no significant change in the overpotential of the electrocatalyst during the 5 h stability test, which makes it an excellent material for large-scale applications.

To assess the dual-purpose nature of the synthesized catalysts, their performance in a HER setup was measured in an alkaline solution using LSV polarization curves at a potential range of 0 to -0.7 V (vs. RHE). Fig. 8(a) represents the HER polarization curves of as-prepared samples, Pt foil and CFC. The current density of -40 mA cm^{-2} and -100 mA cm^{-2} was achieved by $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$, which requires a very low

overpotential of 368 mV and 418 mV than its counterpart $\text{Ni}_2\text{O}_2(\text{OH})$ (471 mV and 519 mV of overpotential at -40 mA cm^{-2} and -100 mA cm^{-2}). While state-of-the-art Pt foil attains a current density of -40 mA cm^{-2} and -100 mA cm^{-2} by requiring an overpotential of 157 mV and 227 mV. The CFC polarization curves demonstrate that they don't get involved during the HER process. The electrocatalytic kinetics can be deduced by analyzing the Tafel slope values (Fig. 8(b)). The calculated Tafel slope (115 mV dec^{-1}) of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ was lesser as compared to the other sample (124 mV dec^{-1}). This indicates that $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ has a favorable process in the HER reaction pathway and excellent intrinsic activity. The Tafel slope of the $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst suggests HER proceeds via the Volmer-Heyrovsky pathway. The durability of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst material was investigated by performing a chronopotentiometry test for 5 h at an overpotential of 368 mV (vs. RHE) for HER electrocatalysis as shown in Fig. 8(d). The chronopotentiometry curve demonstrates a minor change in current density for long time duration which means the electrocatalyst material is efficient and stable for HER electrocatalysis. The results revealed that the design and construction of a CNTs-based- $\text{Ni}_2\text{O}_2(\text{OH})$ electrocatalyst with enormous active sites, faster kinetics, and excellent charge mobility significantly strengthened its HER activity. Moreover, the bifunctional electrocatalytic activity of prepared materials is summarized in Table 2.

EIS analysis was considered to get insight into kinetics and the charge-transfer resistance (R_{ct}). The Nyquist plots of samples were fitted with a modified Randles circuit, as shown in the inset of Fig. 9. The measured R_{ct} values of $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ samples are 5.5Ω and 2.9Ω , respectively. The lower R_{ct} value of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ is significantly lower than that of bare $\text{Ni}_2\text{O}_2(\text{OH})$ revealing fast electron transport for the electrocatalytic reaction. The oxygen vacancy formation in $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ with the tube-like support significantly improves its electrical conductivity, as shown by the R_{ct} comparison with $\text{Ni}_2\text{O}_2(\text{OH})$. The following equation was used to find the exchange current density (j^0) parameter for the materials: $j^0 = RT/nFA (\theta')$, where T denotes absolute temperature, n represents the number of electrons, A refers to the geometric area of the electrode surface, F shows Faraday's constant, and R ascribes the charge transfer resistance of the electrode/electrolyte interphase. The calculated exchange current density of

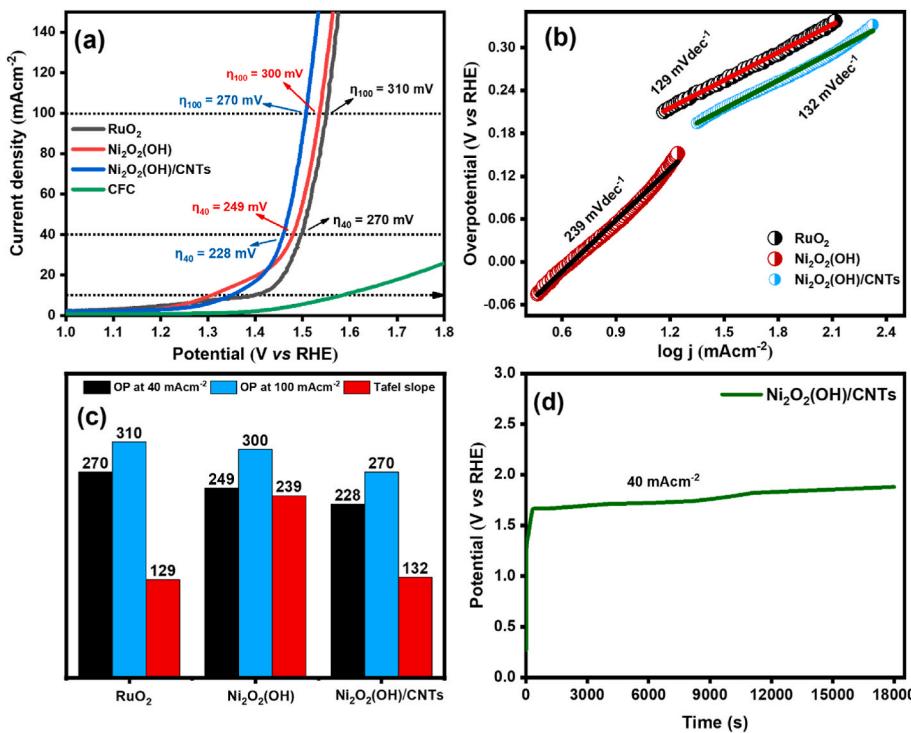


Fig. 7. Representation of OER electrocatalysis results; (a) LSV curves taken at 5 mV/s (b) their corresponding Tafel slope (c) overpotential and Tafel slope bar graphs of RuO₂, Ni₂O₂(OH) and Ni₂O₂(OH)/CNTs electrocatalysts and, (d) Chrono-potentiometric curve of Ni₂O₂(OH)/CNTs electrocatalyst at 40 mA cm⁻² during 5 h of OER electrocatalysis.

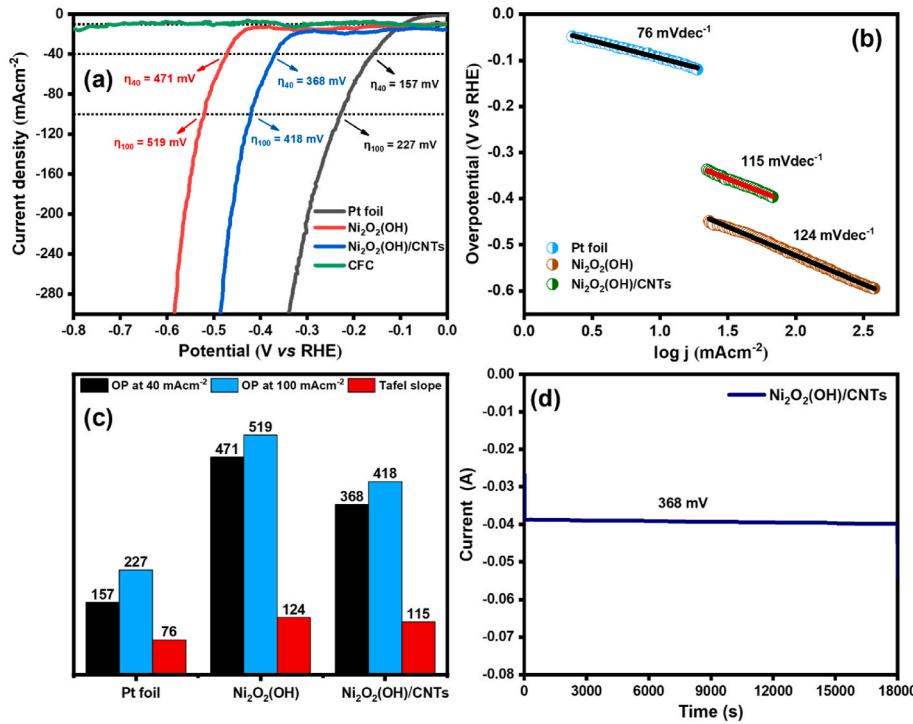


Fig. 8. Illustration of HER electrocatalysis (a) LSV curves taken at 5 mV/s (b) their corresponding Tafel slope plot (c) overpotential and Tafel slope comparison bar graph of as-prepared electrocatalysts and, (d) Chronoamperometric curve of a Ni₂O₂(OH)/CNTs at 368 mV during 5 h of HER electrocatalysis.

Ni₂O₂(OH) and Ni₂O₂(OH)/CNTs is displayed in Table 3. The higher j^0 value of Ni₂O₂(OH)/CNTs is because of its enhanced intrinsic activity for OER and HER and rapid charge transport kinetics as compared to Ni₂O₂(OH).

Mott-Schottky analysis was conducted to evaluate the flat band potential (E_{fb}) and carrier concentration (N_D) of Ni₂O₂(OH) and Ni₂O₂(OH)/CNTs electrocatalysts (Fig. 10). This analysis provides indirect information about the presence of oxygen vacancies. Mott-

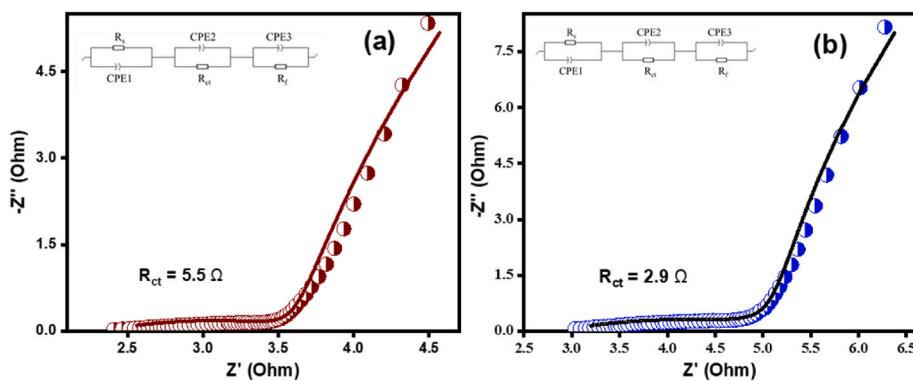


Fig. 9. EIS plots of as-synthesized: (a) bare $\text{Ni}_2\text{O}_2(\text{OH})$ and (b) $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalysts.

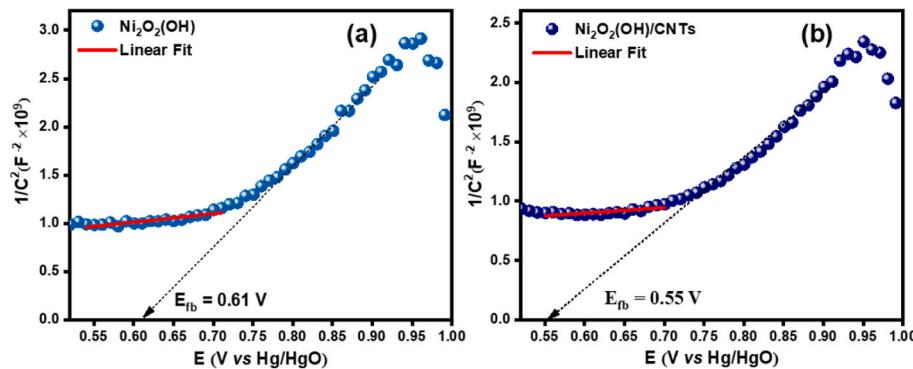


Fig. 10. Mott Schottky analysis of prepared electrocatalysts; (a) $\text{Ni}_2\text{O}_2(\text{OH})$ and, (b) $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$.

Schottky tests were considered to observe the electrical properties of the solid-liquid interface. The E_{fb} of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ was calculated to be 0.55 V, which is significantly lower as compared to $\text{Ni}_2\text{O}_2(\text{OH})$. The shift towards a lower potential suggests that the presence of CNTs in the composite affects the electronic properties of $\text{Ni}_2\text{O}_2(\text{OH})$, potentially altering its catalytic activity for water splitting. Furthermore, another important parameter, N_D of prepared samples has been calculated for the assessment of oxygen vacancies. The $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ catalyst has a higher calculated carrier concentration compared to $\text{Ni}_2\text{O}_2(\text{OH})$ as presented in Table 3. This suggests that the presence of CNTs in the composite enhances the formation of oxygen vacancies or promotes electron transfer processes. The increased carrier concentration implies an improved charge transport capability, which can be beneficial for catalytic reactions such as water splitting. Oxygen vacancies in metal oxide materials, such as $\text{Ni}_2\text{O}_2(\text{OH})$, are lattice defects where oxygen atoms are missing from the crystal structure. These vacancies can act as

active sites for electrochemical reactions, including water splitting. The higher carrier concentration in $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ indicates a higher density of oxygen vacancies, likely due to the presence of CNTs, which promotes defect formation. The reduced flat band potential and higher carrier concentration in $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ suggest that the presence of CNTs affects the electronic properties and may enhance the catalytic activity for water splitting, potentially impacting electrochemical water splitting reactions (OER/HER). Moreover, the comparison of the current study with the different electrocatalysts synthesized by various strategies from the literature is shown in Table 4.

4. Conclusions

Herein, $\text{Ni}_2\text{O}_2(\text{OH})$ and $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalysts were synthesized using an easy, simple, and economical ultrasonication method. The prepared material's electrochemical performance was assessed by

Table 4

Comparative studies of the bifunctional $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ electrocatalyst with previously reported literature.

Electrocatalysts	Electrolyte	Method of synthesis	Overpotential (mV)		Tafel slope (mVdec ⁻¹)		Ref.
			OER	HER	OER	HER	
$\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$	1 M KOH	Ultrasonication route	$\eta_{40} = 228$ $\eta_{100} = 270$	$\eta_{40} = 368$ $\eta_{100} = 418$	132	115	This work
$\beta\text{-Ni(OH)}_2/\text{NF}$	1.0 M KOH	Facile hydrothermal etching method	$\eta_{50} = 329$	$\eta_{50} = 170$	40	51	[53]
$\text{Ni}/\text{NiO}@\text{rGO}$	0.5 M H_2SO_4	Chemical method	$\eta_{10} = 480$	$\eta_{10} = 582$	41	63	[54]
$\text{Ni(OH)}_2/\text{NiCo}_2\text{O}_4$	1 M KOH	One-step hydrothermal	$\eta_{10} = 224$	$\eta_{10} = 189$	43	41	[55]
$\text{S-Ni(Fe)}\text{OOH}$	1 M KOH	Chemical method	198	22	17.50	93.21	[56]
Ni-Ni(OH)_2	1.0 M KOH	Electrodeposition approach	–	$\eta_{10} = 73$ $\eta_{100} = 336$	–	33	[57]
$\text{NF-Ni(OH)}_2\text{-Ni}^{\text{vac}}\text{Fe-LDHs}$	1 M KOH	Two-step hydrothermal method	$\eta_{200} = 376$	$\eta_{200} = 247$	121.37	95.87	[58]
$\text{Ni}_3\text{S}_2/\text{Ni(OH)}_2$	1 M KOH	Electrodeposition	$\eta_{20} = 249$	$\eta_{10} = 66$	61	59	[59]
$\text{Mo-Ni(OH)}_2/\text{Fe}_x\text{Ni}_y\text{(OH)}_{3x+y}$	1 M KOH	Hydrothermal method	$\eta_{10} = 229$	$\eta_{10} = 57$	33.53	75.31	[60]
NiO nanosheets	1 M KOH	Hydrothermal method	380	83	299	209	[61]

using the electrochemical workstation. Electrocatalytic materials synthesized via the ultrasonication excitation pathway are exceptional for use in the process of water splitting (OER/HER) due to their high electrochemically active surface area of 4.1 cm^2 for $\text{Ni}_2\text{O}_2(\text{OH})$ and 6 cm^2 for $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$. When CNTs were introduced into the as-prepared $\text{Ni}_2\text{O}_2(\text{OH})$, the water-splitting ability of the resulting $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ nanocomposite was effectively enhanced as they affected the electronic properties of the catalyst. The lowest overpotential (228 mV; 270 mV and 368 mV; 418 mV for OER and HER, respectively, at current density 40 and 100 mA cm^{-2}), minimum tafel slope, reduced charge transfer resistance value (2.9Ω), lesser flat band potential (0.55 V), excellent exchange current density (2.2 mA cm^{-2}), and higher carrier concentration ($14.7 \times 10^{22} \text{ cm}^{-3}$) reveal the superior bifunctional water splitting performance of $\text{Ni}_2\text{O}_2(\text{OH})/\text{CNTs}$ material as compared to $\text{Ni}_2\text{O}_2(\text{OH})$. Its excellent performance can be attributed to an exceptionally conductive substrate network provided by carbon-based materials (CNTs), which are effective for introducing more active sites, excellent electron mobility, and a greater exposed surface area.

Declaration of Competing interest

No conflict of interest

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