

1 **Anomalous Tunneling Magnetoresistance Oscillation and Electrically Tunable**
2 **Tunneling Anisotropic Magnetoresistance in Few-layer CrPS₄**
3

4 *ZhuangEn Fu, Hong-Fei Huang, Piumi Samarawickrama, Kenji Watanabe, Takashi*
5 *Taniguchi, Wenyong Wang, John Ackerman, Jiadong Zang, Jie-Xiang Yu*, and Jifa Tian**

6

7 Z. Fu, P. Samarawickrama, W. Wang, J. Tian
8 Department of Physics and Astronomy, University of Wyoming, Laramie, Wyoming 82071,
9 United States
10 E-mail: jtian@uwyo.edu

11

12 Z. Fu, P. Samarawickrama, J. Tian
13 Center for Quantum Information Science and Engineering, University of Wyoming, Laramie,
14 Wyoming 82071, United States

15

16 H. Huang, J. Yu
17 School of Physical Science and Technology, Soochow University, Suzhou 215006, China
18 E-mail: jxy@suda.edu.cn

19

20 K. Watanabe, T. Taniguchi
21 Research Center for Electronic and Optical Materials, National Institute for Materials Science,
22 1-1 Namiki, Tsukuba 305-0044, Japan

23

24 K. Watanabe, T. Taniguchi
25 Research Center for Materials Nanoarchitectonics, National Institute for Materials Science, 1-
26 1 Namiki, Tsukuba 305-0044, Japan

27

28 J. Ackerman
29 Department of Chemical and Biomedical Engineering, University of Wyoming, Laramie,
30 Wyoming 82071, United States

31

32 J. Zang
33 Department of Physics and Astronomy, University of New Hampshire, Durham, New
34 Hampshire 03824, United States

35

36

37

38

39 Keywords: 2D magnets, CrPS₄, tunneling anisotropic magnetoresistance, magnetoresistance
40 oscillation, magnetic tunneling junction.

41

42

43

44

45

1 ABSTRACT

2 Two-dimensional (2D) van der Waals (vdW) magnets with layer-dependent magnetic states
3 and/or diverse magnetic interactions and anisotropies have attracted extensive research interest.
4 Despite the advances, a notable challenge persists in effectively manipulating the tunneling
5 anisotropic magnetoresistance (TAMR) of 2D vdW magnet-based magnetic tunnel junctions
6 (MTJs). Here, we report the novel and anomalous tunneling magnetoresistance (TMR)
7 oscillations and pioneering demonstration of bias and gate voltage controllable TAMR in 2D
8 vdW MTJs, utilizing few-layer CrPS₄. This material, inherently an antiferromagnet, transitions
9 to a canted magnetic order upon application of external magnetic fields. Through TMR
10 measurements, we unveil the novel, layer-dependent oscillations in the tunneling resistance for
11 few-layer CrPS₄ devices under both out-of-plane and in-plane magnetic fields, with a
12 pronounced controllability via gate voltage. Intriguingly, we demonstrate that both the polarity
13 and magnitude of TAMR in CrPS₄ can be effectively tuned through either a bias or gate voltage.
14 We further elucidate the mechanism behind this electrically tunable TAMR through first-
15 principles calculations. The implications of our findings are far-reaching, providing new
16 insights into 2D magnetism and opening avenues for the development of innovative spintronic
17 devices based on 2D vdW magnets.

18

19 **1. Introduction**

20 Since 2017, the study of two-dimensional (2D) van der Waals (vdW) magnets has become a
21 key area of scientific and technological importance^[1,2]. The 2D vdW magnets featuring
22 distinctive properties, especially their marked layer-dependent magnetism and diverse types of
23 magnetic interaction and anisotropy^[1–25], offer ideal platforms for exploring magnetism down
24 to 2D limit. In the past few years, different approaches/techniques have been developed to
25 identify and manipulate magnetism and magnetic interactions of atomically thin 2D vdW
26 magnets, including magneto-optic Kerr microscopy^[1,2], single-spin microscopy^[26], Raman
27 spectroscopy^[27], second-harmonic generation^[28], spin-polarized scanning tunneling
28 microscopy^[11], electrostatic doping^[6–8], pressure^[29], circularly polarized light^[30], tunneling
29 transport^[31–37], etc. Among them, an exciting development is the emergence of the 2D vdW
30 magnet-based magnetic tunnel junctions (MTJs). Different from conventional MTJs composed
31 of a structure of a ferromagnet (FM)/insulating spacer/FM, the 2D vdW-based MTJs utilize
32 atomically thin 2D insulating or semiconducting magnetic layers that function both as a spin-
33 polarized current generator and a tunneling barrier. This innovative architecture adds a new
34 dimension to MTJs, leveraging the intrinsic characteristics of 2D magnets. Novel magnetic

1 properties and quantum phenomena of 2D vdW magnets can be explored through tunneling
2 measurements, harnessing the potential of next-generation spintronic devices. A notable
3 example is graphene/CrI₃/graphene^[31,33–37] MTJ devices. In the past few years, novel properties
4 and device prototypes of 2D vdW-based MTJs have been explored using tunneling transport,
5 for instance, layer-dependent 2D magnetism in CrX₃^[31,35,38], spin tunnel field-effect transistors
6 in CrI₃^[33], giant magnetoresistance in CrI₃^[34,39], magnon-assisted tunneling in CrBr₃^[40],
7 magneto-memristive effects in CrI₃^[36,41], etc. While considerable advancements have been
8 achieved in the field, the majority of research has centered on few-layer CrI₃-based MTJs,
9 which are recognized for their layer-dependent magnetism, strong magnetic anisotropy, and
10 spin-flip transition characteristics. However, considering the wide array of magnetic
11 interactions and anisotropies, it is essential to explore a more diverse range of magnetic
12 states/interactions, such as spin-flop transition and tunability of magnetic anisotropy in 2D
13 semiconducting vdW magnets.

14 Chromium thiophosphate (CrPS₄)^[3,42–56] represents a paradigmatic example, exemplifying the
15 intricate interplay between its magnetic, structural, electrical, and optical properties. For
16 instance, in monolayer CrPS₄, quasi-1D chains of edge-sharing CrS₆ octahedra extend along
17 the *b*-axis, interconnected along the *a*-axis by PS₄ tetrahedra^[3,42]. Bulk CrPS₄ exhibits intralayer
18 ferromagnetic coupling and interlayer antiferromagnetic, classified as an A-type
19 antiferromagnet with a Néel temperature of ~ 38 K^[50]. Contrary to CrI₃, CrPS₄ not only
20 demonstrates stability under ambient conditions^[48], but also features spin canting in individual
21 layers under an external magnetic field (**Figure 1**), leading to a spin-flop phase transition^[57,60].
22 Furthermore, CrPS₄ is a magnetic semiconductor with an indirect band gap of ~ 1.4 or 1.31 eV
23 determined by optical measurements^[3,57], offering a unique opportunity for tuning their
24 electronic and magnetic properties via gate voltages^[43]. Despite growing interest in exploring
25 the unique properties of CrPS₄, aspects such as its layer-dependent magnetism, quantum effects,
26 and the potential for tunable magnetic anisotropy in its few-layer form are still largely
27 unexplored.

28 In this work, based on tunneling magnetoresistance (TMR) measurement, we report layer-
29 dependent characteristics of TMR in few-layer CrPS₄-based MTJs. We find that the TMR of
30 the few-layer CrPS₄ shows anomalous and novel TMR oscillations with varying out-of-plane
31 (OOP) or in-plane (IP) magnetic fields, depending on the number of layers. The TMR is further
32 found to be highly tunable by electrostatic doping. Strikingly, we demonstrate that the polarity
33 and magnitude of tunneling anisotropic magnetoresistance (TAMR) in few-layer CrPS₄ can be
34 controlled by either a bias or a gate voltage. Lastly, we employ first-principles calculations to

1 elucidate the underlying mechanism of the electrically tunable TAMR in few-layer CrPS₄. Our
2 findings not only contribute to the fundamental understanding of 2D magnetism and novel
3 quantum effects in vdW magnetic materials but also pave the way for developing ultra-compact,
4 energy-efficient spintronic devices, thereby marking a significant stride in the manipulation and
5 utilization of spin states in low-dimensional systems.

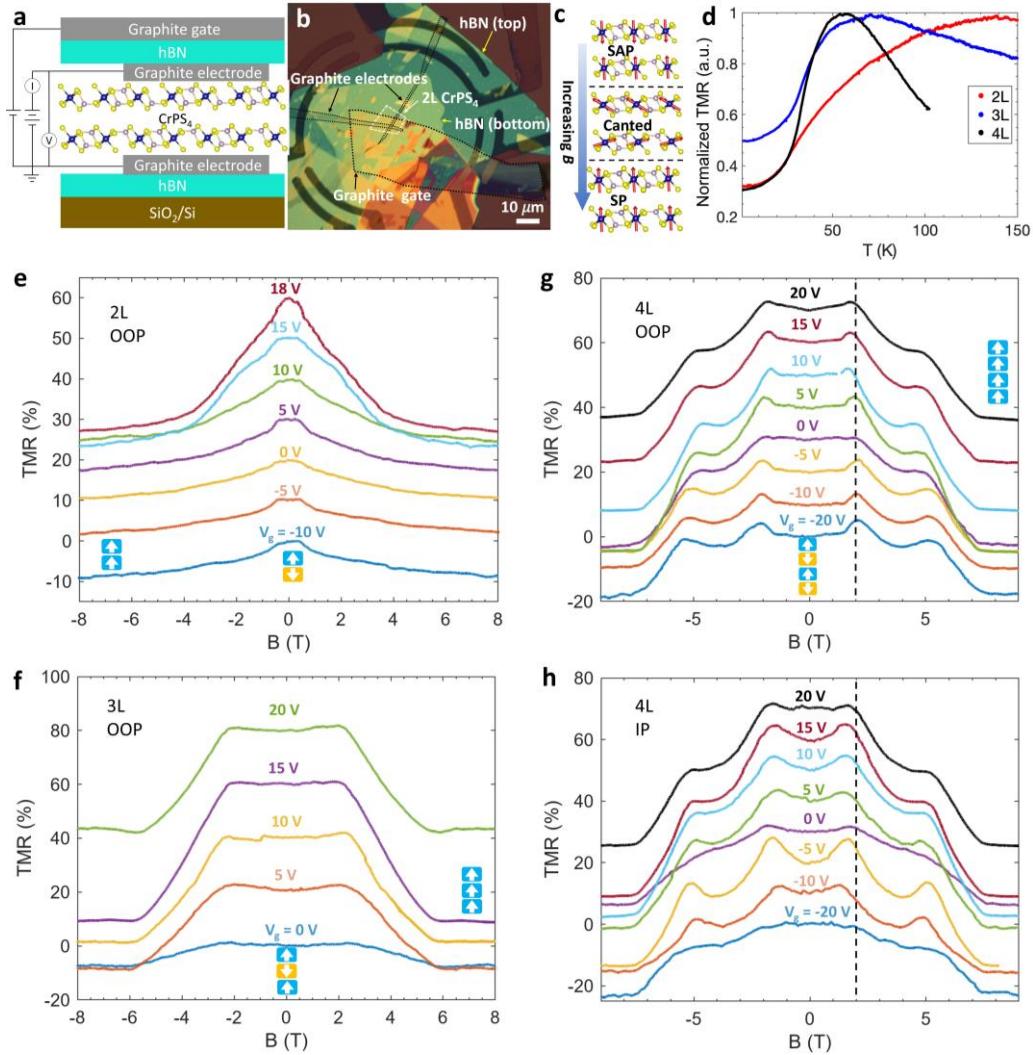
6 **2. Results**

7 **2.1. Layer- and Gate-dependent Tunneling Magnetotransport in Few-layer CrPS₄**

8 To probe the layer-dependent magnetic states in few-layer CrPS₄, we employed tunneling
9 transport measurements^[31,32,39]. The CrPS₄-based MTJs with both top and bottom gates were
10 fabricated by the commonly used dry transfer method (see Methods). **Figure 1a** and **1b** show
11 a schematic side view and an optical image of a bilayer (2L) CrPS₄-based MTJ, respectively.
12 This device features a structure of graphite (top gate)/hBN/graphite (electrode)/CrPS₄/graphite
13 (electrode)/hBN on a SiO₂ (285 nm)/Si substrate (back gate). The high quality of the CrPS₄-
14 based tunnel junction is evidenced by the exponential increase in current with increasing bias
15 voltage, consistently observed in the presence and absence of magnetic fields (see **Figure S1**
16 in Supporting Informatino). It is known that bulk CrPS₄ is an A-type antiferromagnet and
17 undergoes a spin-flop transition from a spin-antiparallel (SAP) state to a canted spin state at B
18 ≈ 0.7 T, followed by a transition to a spin-parallel (SP) state at $B \approx 8.0$ T,^[50] as schematically
19 illustrated in **Figure 1c**. **Figure 1d** shows the tunneling resistance as a function of temperature
20 (R vs. T) for few-layer CrPS₄ devices with different thicknesses, including bilayer (2L, ~ 1.2
21 nm), trilayer (3L, ~ 1.8 nm) and quadra-layer (4L, ~ 2.4 nm). We see that for all the CrPS₄
22 devices, as the temperature decreases, the tunneling resistance initially increases, followed by
23 a subsequent decrease. In stark contrast to few-layer CrI₃ tunnel devices^[39], the decrease of the
24 tunneling resistance of the CrPS₄ samples at low temperatures suggests a diminished influence
25 of the spin filtering effect, typically associated with interlayer AFM ordering in few-layer 2D
26 magnets. Furthermore, in CrI₃, the pronounced increase in tunneling resistance correlates
27 directly with the T_N , serving as a characteristic marker. However, this correlation does not hold
28 for few-layer CrPS₄. As depicted in **Figure 1d**, the downturns in resistance occur at ~ 120 K
29 for 2L CrPS₄, ~ 70 K for 3L, and ~ 55 K for 4L samples. Intriguingly, these temperatures
30 significantly exceed the T_N of 38 K for bulk CrPS₄, suggesting alternative mechanisms (such as
31 metal-semiconductor/insulator transition or temperature induced distortion in crystal structure)
32 influencing the thermal behavior of the tunneling resistance in these CrPS₄ thin layers.

33

34



1 **Figure 1.** Layer- and gate-dependent magnetism in few-layer CrPS₄. a) Schematic of the structure of a
2 CrPS₄ tunneling device with a stack of graphite/hBN/graphite /CrPS₄/graphite/hBN. b) Optical image
3 of the top view of the 2L CrPS₄ tunneling device as illustrated in (a). c) Schematic for the evolution of
4 spin structure under various magnetic field in CrPS₄. Spin-antiparallel (SAP), canted spin (Canted), and
5 spin-parallel (SP) correspond to magnetic structure at low, intermediate, and high magnetic field,
6 respectively. d) Temperature-dependent normalized tunneling resistance R for 2L (red), 3L (blue) and
7 4L (black) CrPS₄ tunneling devices. e-g) TMR vs. B at different gate voltages for 2L (e), 3L (f) and 4L
8 (g) CrPS₄ tunneling devices. The bias voltages applied were -450, 700 and -600 mV for 2L, 3L and 4L
9 CrPS₄, respectively. The arrows indicate the magnetization of the corresponding CrPS₄ layers at given
10 magnetic fields. The dash line in (g) marks $B = 2$ T. The applied magnetic field is along out-of-plane
11 (OOP) direction. h) TMR as a function of gate voltage for the 4L CrPS₄ under an in-plane (IP) magnetic
12 field. The dash line marks $B = 2$ T. The measurements in (e)- (h) were performed at $T = 1.5$ K. The TMR
13 curves have been vertically shifted to enhance clarity.

14

15 We then explore the dependence of the TMR on the gate voltage for CrPS₄-based MTJs under
16 both OOP and IP magnetic fields. The TMR is defined as $\text{TMR} = (R_B - R_{B=0})/R_{B=0} \times 100\%$,

1 where R_B is the tunneling resistance at a given magnetic field B and $R_{B=0}$ is tunneling
 2 resistance measured at $B = 0$ T. **Figure S2** (Supporting Information) shows the gate-dependent
 3 TMR for three CrPS₄-based MTJs under different OOP magnetic fields. For the 2L CrPS₄
 4 device, the TMR generally becomes more negative as the gate voltage increases at different
 5 magnetic fields. For instance, at $B = 7$ T, the TMR drops from $\sim -10\%$ to $\sim -32\%$ as the gate
 6 voltage increases from 0 V to 18 V. In the case of the 3L CrPS₄ device, the TMR initially
 7 decreases with the increasing gate voltage, reaching a minimum (maximum in magnitude) at
 8 around 15 V, followed by an increase upon further voltage elevation. For instance, at $B = 8$ T,
 9 there is a significant decrease in TMR from roughly -10% to -50% as the gate voltage extends
 10 from 0 V to 15 V. For the 4L CrPS₄, a similar trend is observed, where the TMR first decreases
 11 with an increasing V_g , peaking at $V_g \sim 15$ V, before increasing with subsequent increases in gate
 12 voltage. We note that the maximum gate voltages for the 2L, 3L, and 4L CrPS₄ tunneling
 13 devices, defined as the thresholds beyond which significant leakage current (less than 50 pA)
 14 is observed, are 18 V, 20 V, and 20 V, respectively.

15 **Figure 1e-g** show the corresponding TMR as a function of the OOP magnetic field under
 16 different gate voltages for 2L, 3L and 4L CrPS₄, respectively. For the 2L CrPS₄, depicted in
 17 **Figure 1e**, the ground state at zero magnetic field is an antiferromagnetic (AFM) state denoted
 18 as $\uparrow\downarrow$, where each arrow represents the spin orientation in the corresponding layer. As the
 19 magnetic field increases, a notable decrease in TMR is observed when the magnitude of B
 20 approaches ~ 0.4 T. The gradual decrease of TMR indicates a spin-flop transition from the AFM
 21 to a canted spin state (**Figure 1c**), in contrast to a sudden change of TMR in few-layer CrI₃ due
 22 to the spin-flip transition^[32,37]. As the magnitude of B increases further, the TMR continues to
 23 decline and starts to level off (e.g., at $V_g = 10, 15$ and 18 V), signifying a transition from the
 24 canted spin state to a fully SP state ($\uparrow\uparrow$). Importantly, we reveal a strong dependence of TMR
 25 on gate voltage. When the gate voltage is increased from -10 to 18 V, two distinct TMR
 26 characteristics emerge: 1) a significant decrease in the TMR (being more negative), and 2) the
 27 gradual decrease and onset of saturation, clearly indicating a spin canted region and the fully
 28 spin-polarized magnetic state. The 3L CrPS₄ device, illustrated in **Figure 1f**, exhibits a ground
 29 state characterized by interlayer AFM coupling, denoted as either $\uparrow\downarrow\uparrow$ or $\downarrow\uparrow\downarrow$. Unlike the 2L
 30 CrPS₄, when subjected to increasing magnetic fields, the TMR of the 3L sample initially shows
 31 a slight increase until the magnetic field reaches ~ 2.2 T. Beyond this point, TMR rapidly
 32 decreases as B continues to increase, ultimately reaching saturation at ~ 5.8 T for all the gate
 33 voltages. This behavior indicates a distinct phase transition from the $\uparrow\downarrow\uparrow$ (or $\downarrow\uparrow\downarrow$) state to a
 34 fully SP ($\uparrow\uparrow\uparrow$) state through the spin-flop process. In terms of gate voltage dependence, the

1 TMR response of the 3L CrPS₄ is also notable. As the V_g increases from 0 to 20 V, the difference
2 in TMR magnitude between $B = 0$ and 9 T initially grows, peaking at $V_g = 15$ V, before
3 subsequently diminishing. In the 4L CrPS₄ device, we observed a series of more intriguing
4 features. **Figure 1g** and **1h** show the 4L sample TMR as a function of OOP and IP magnetic
5 fields, respectively, at various gate voltages. Specifically, under most gate voltages, the TMR
6 in 4L CrPS₄ exhibits a progressive increase with increasing OOP magnetic fields, culminating
7 in a pronounced peak around 2 T. This contrasts markedly with the behavior observed in the
8 few-layer CrI₃ tunneling device, where TMR, predominantly driven by the spin filtering effect,
9 remains largely invariant until the onset of a spin-flip transition. Furthermore, the 4L CrPS₄
10 undergoes a spin-flop transition, evolving into the fully SP state ($\uparrow\uparrow\uparrow\uparrow$) at approximately 7.1
11 T. Notably, amidst this transition, we detected an additional TMR peak around 5 T for all
12 applied gate voltages. This finding is paralleled by the observation of similar peak-like TMR
13 features under IP magnetic fields (**Figure 1h**). Additionally, our data indicate that the gate
14 voltage has a substantial influence on the TMR characteristics in the 4L CrPS₄ device,
15 significantly affecting both the magnitude and the peak positions (black dashed lines) of the
16 TMR curves under both OOP and IP magnetic fields.

17 Next, we study the intriguing peak-like features observed in the 4L CrPS₄ samples. Our
18 transport results indicate these features are weak in the 2L and 3L samples. Considering the
19 structural similarity across all devices and the occurrence of these peak-like features in 4L
20 CrPS₄ under both OOP and IP magnetic fields, we can confidently discount the Shubnikov-de
21 Haas oscillations, potentially associated with few-layer graphene electrodes, as a possible cause.
22 Further, these peak-like features are discernible under both OOP and IP magnetic fields, with
23 no obvious shift of the magnetic fields at the corresponding TMR peak positions for the two
24 magnetic field orientations. Such observations lead us to speculate that these features are
25 unlikely to be associated with the emergence of intermediate, layer-dependent spin
26 configurations, such as that observed in few-layer CrI₃. This is underpinned by the fact that the
27 magnetic structure of CrPS₄ predominantly exhibits an easy axis along the OOP direction,
28 which would necessitate a substantially stronger IP magnetic field to effectuate a change in the
29 spin state. Also, since CrPS₄ undergoes a spin-flop transition instead of a spin-flip transition,
30 the characteristic peaks and valleys of TMR do not align with the spin-filtering effect.
31 Consequently, we propose that these peak-like features could be manifestations of magnetic
32 field-induced oscillations in TMR of few-layer CrPS₄. We note that these TMR oscillations
33 have not been observed in other 2D magnets, thus far. A plausible explanation for the TMR
34 oscillations could be the spin geometric phase mechanism between two spin current tunneling

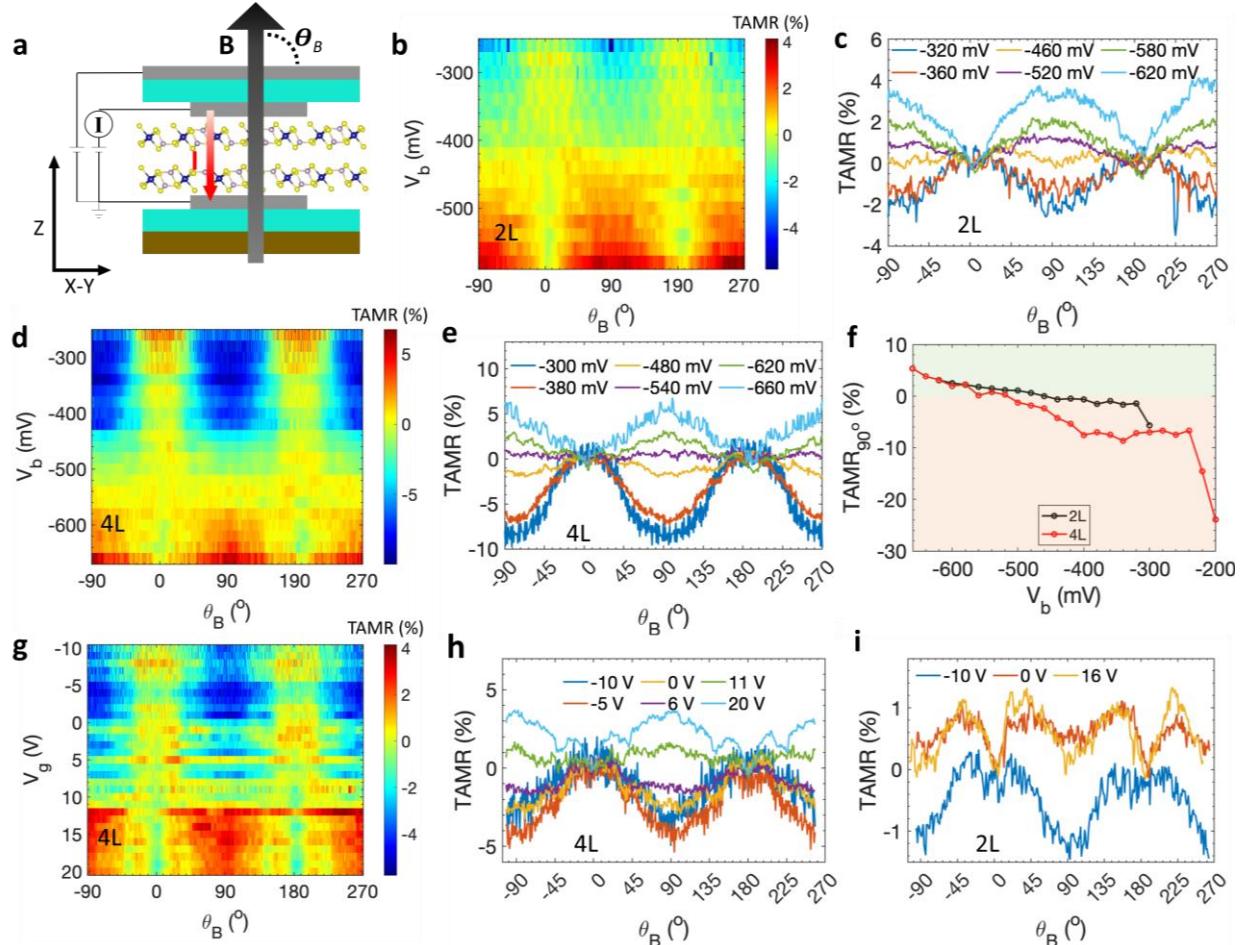
1 channels, where the tunneling electron spin interacts with the canted spin textures in both
 2 channels. Variations in magnetic field flux and the number of layers are likely to induce changes
 3 in the spin geometric phase^[58–61], resulting in oscillations of the tunneling current. However, a
 4 comprehensive understanding of these observed oscillations necessitates further experimental
 5 and theoretical investigations. We further note that the measured TMR of few-layer CrPS₄
 6 encompasses both the spin-flop transition and TMR oscillations induced by the magnetic field.
 7 The distinct variations in TMR observed in the CrPS₄ tunnel junction devices with different
 8 layers can be attributed to the following factors. Firstly, it is documented that 2D magnets often
 9 exhibit layer-dependent coercivity for TMR transitions particularly in the few-layer region, as
 10 evidenced in other 2D magnetic systems like CrI₃. Secondly, our results indicate that TMR
 11 oscillations are more pronounced in thicker layers.

12 **2.2. Electrically Tunable Tunneling Anisotropic Magnetoresistance**

13 It is well-accepted that anisotropic magnetoresistance (AMR), which describes how the
 14 magnetoresistance changes with magnetic field direction in a ferromagnetic conducting film,
 15 leads to the relation: $\rho(\theta) = \rho_{\perp} + (\rho_{\parallel} - \rho_{\perp})\cos^2\theta$, where ρ is the resistivity of the film, θ is
 16 the angle between the direction of magnetization and current, and ρ_{\parallel} and ρ_{\perp} are resistivities at
 17 $\theta = 0^\circ$ and $\theta = 90^\circ$, respectively. The origin of AMR can be traced back to spin-orbital
 18 coupling. When considering the tunneling transport, a more complex bias-dependent TAMR
 19 was observed in conventional MTJs^[62] such as CoFe/MgO/CoFe and CoFe/Al₂O₃/CoFe.
 20 However, TAMR in 2D magnet-based MTJs remains elusive. In the few-layer CrPS₄-based
 21 MTJs, we investigate the influence of three key parameters on the TMR: 1) the bias voltage and
 22 2) gate voltage applied to the junction as well as 3) the angle between the external magnetic
 23 field and the IP direction of the few-layer CrPS₄.

24 We find that both the polarity and magnitude of the TAMR in these MTJs can be extensively
 25 modulated through electrical methods. Here, we define the angle θ_B as the angle between the B
 26 field and the sample plane, as shown in **Figure 2a**. The TAMR is calculated using the formula
 27 $\text{TAMR} = (R_{\theta_B} - R_{\theta_B=0^\circ}) \times 100\% / R_{\theta_B=0^\circ}$, where R_{θ_B} is the tunneling resistance at a given
 28 magnetic field and $R_{\theta_B=0^\circ}$ represents the tunneling resistance at $\theta_B = 0^\circ$. **Figure 2b** shows a
 29 colormap depicting the TAMR of the 2L CrPS₄ MTJ as a function of θ_B under various bias
 30 voltages at $B = 7$ T and $T = 1.5$ K. We note that the TAMR is $\sim 0\%$ at both $\theta_B = 0^\circ$ and 180°
 31 and the corresponding magnetic states of the few-layer CrPS₄ are in the fully SP state. **Figure**
 32 **2c** exhibits the line cuts of TAMRs for selected biases, as derived from Figure 2b. We find that
 33 the TAMR has a two-fold rotational symmetry regardless of the applied bias, as shown Figure

1 2b and 2c. This behavior can be primarily attributed to the even symmetry typically observed
 2 in the TMR of few-layer 2D magnets relative to the applied magnetic field. Remarkably, as the
 3 bias voltage varies from -620 to -320 mV, we see that the TAMR of 2L CrPS₄ undergoes a
 4 change in magnitude and a sign reversal. For instance, at a bias voltage (e.g., -320 mV), the
 5 TAMR is negative, displaying peaks at $\theta_B = 0^\circ$ and 180° , while the valleys occur at $\theta_B = 90^\circ$
 6 and 270° . Conversely, at a more negative bias (- 620 mV), the TAMR becomes positive, with
 7 valleys at $\theta_B = 0^\circ$ and 180° , and peaks at $\theta_B = 90^\circ$ and 270° . Consequently, the TAMR at $\theta_B =$
 8 90° increases from -2% to 4% when the bias voltage changes from -320 mV to -620 mV. We
 9 further demonstrate that the bias-voltage tunable TAMR can also be realized in the 4L CrPS₄.



11 **Figure 2.** Bias- and gate-dependent TAMR in few-layer CrPS₄. a) Schematic for the definition of angle
 12 of B field θ_B . The black arrow indicates the B field direction. The red arrow indicates the current direction.
 13 b) TAMR as a function of θ_B at different bias voltages for the 2L CrPS₄ tunneling device. The
 14 measurements were performed at $T = 1.5$ K. The applied B field is 7 T. c) TAMR as a function of θ_B at
 15 selected bias voltage V_b taken from (b). d) TAMR as a function of θ_B at different bias voltages for the
 16 4L CrPS₄ tunneling device. The measurements were performed at $T = 1.5$ K. The applied B field was 9
 17 T. e) TAMR as a function of θ_B at selected bias voltage V_b 's taken from (d). f) TAMR at $\theta_B = 90^\circ$ as a
 18 function of bias voltage for the 2L and 4L CrPS₄ tunneling devices. The data are taken from (b) and (d).

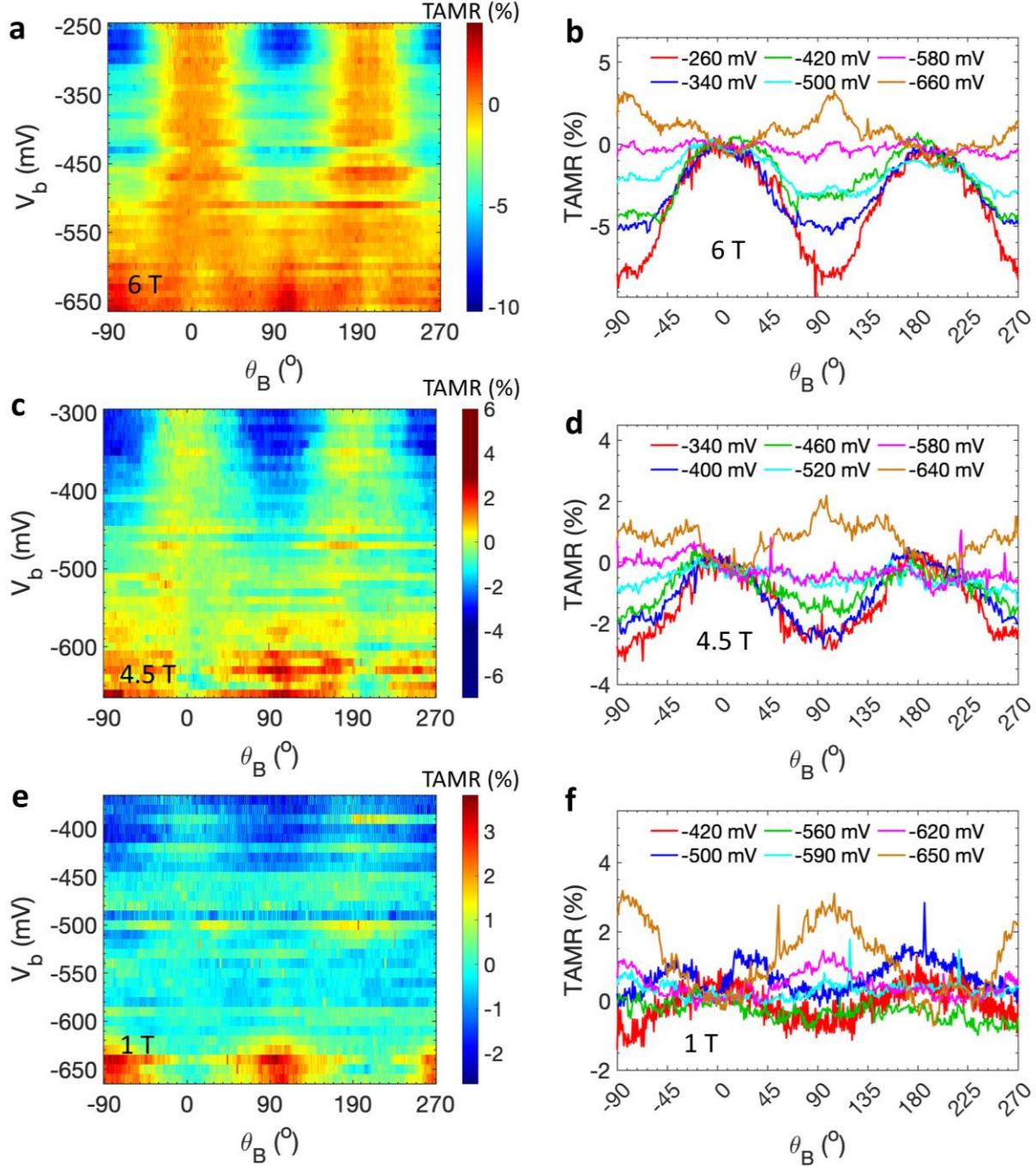
1 g) TAMR as a function of θ_B at different gate voltages V_g for the 4L CrPS₄ tunneling device. The
 2 measurements were performed at $T = 1.5$ K. The applied B field was 9 T. h) TAMR as a function of θ_B
 3 at selected gate voltage V_g 's taken from (g). i) TAMR as a function of θ_B at selected gate voltage V_g 's
 4 for the 2L CrPS₄ tunneling device at $T = 1.5$ K and $B = 7$ T.

5
 6 **Figure 2d** shows the corresponding colormap of the TAMR vs. θ_B under different bias voltages
 7 measured at $T = 1.5$ K and $B = 9$ T. **Figure 2e** shows the line cuts of TAMRs taken from Figure
 8 2d for selected biases. **Figure 2f** displays the TAMR for both the 2L (black dots) and 4L (red
 9 dots) CrPS₄ tunneling devices, measured at $\theta_B = 90^\circ$, and plotted as a function of bias voltage.
 10 We see that the TAMR decreases as the magnitude of the bias voltages reduces. The critical
 11 bias voltages for sign reversal are about -420 mV and -520 mV for 2L and 4L CrPS₄ MTJs,
 12 respectively. We note that a similar bias voltage-tunable TAMR behavior can also be observed
 13 in 3L CrPS₄ (see **Figure S3** in Supporting Information) and an extended positive bias range in
 14 the 4L CrPS₄ (see **Figure S4** in Supporting Information), suggesting that this phenomenon is a
 15 universal characteristic of few-layer CrPS₄-based MTJs.

16 We further explore the dependences of both the polarity and magnitude of TAMR on the applied
 17 gate voltages. **Figure 2g** is the colormap that depicts the TAMR of the 4L CrPS₄ device as a
 18 function of θ_B under various gate voltages at a fixed B field of 9 T and $T = 1.5$ K. The line cuts
 19 corresponding to selected gate voltages are shown in **Figure 2h**. We observe that as the gate
 20 voltage changes from negative to positive, the sign of the TAMR transitions from negative to
 21 positive, exhibiting a reversal behavior similar to that observed under changes in the bias
 22 voltage. For the 2L CrPS₄ tunneling device, we see that the gate voltage can also tune the
 23 magnitude and the sign of TAMR, as shown in **Figure 2i**. However, the corresponding gate
 24 voltage effect on the TAMR in 2L CrPS₄ seems weaker than that of the 4L sample. In general,
 25 we conclude that gate and bias voltages play a similar role in controlling the polarity and
 26 magnitude of TAMR in few-layer CrPS₄-based MTJs. We also notice that, compared to the 4L
 27 CrPS₄, the distorted gate voltage dependence of TAMR in the 2L CrPS₄ device (**Figure 2i**)
 28 suggests a possible doping-induced alteration in the crystal structure and hence the magnetic
 29 ground state. In addition, upon comparison with **Figure 1**, it becomes evident that the data
 30 presented in **Figure 2** displays higher noise levels and noticeable sharp peaks. This discrepancy
 31 can be attributed to the significantly smaller magnitude of the TAMR observed in **Figure 2**,
 32 leading to the amplified noise levels in the curves.

33 The electrically tunable TAMR in the 4L CrPS₄ has also been observed in the SP phase region
 34 and the magnetic states (such as a canted spin state and an AFM state) under considerably lower

1 magnetic fields. **Figure 3a** shows the corresponding colormap of TAMR versus θ_B under
 2 different bias voltages measured at $B = 6$ T, a condition that places the 4L CrPS₄ in a canted
 3 spin phase (as depicted in Figure 1g). The behavior of TAMR vs. θ_B under selected bias voltages



5 **Figure 3.** Bias-dependent polarity of TAMR in the 4L CrPS₄ tunneling device at $B = 6, 4.5$ and 1 T. a)
 6 TAMR as a function of θ_B at different bias voltages for the 4L CrPS₄ tunneling device. The
 7 measurements were performed at $T = 1.5$ K. The applied B field was 6 T. b) TAMR as a function of θ_B
 8 at selected bias voltage V_b 's taken from (a). c) The same as (a) but the applied B field is 4.5 T. d) TAMR
 9 as a function of θ_B at selected bias voltage V_b 's taken from (c). e) The same as (a) but the applied B field
 10 is 1 T. f) TAMR as a function of θ_B at selected bias voltage V_b 's taken from (e).

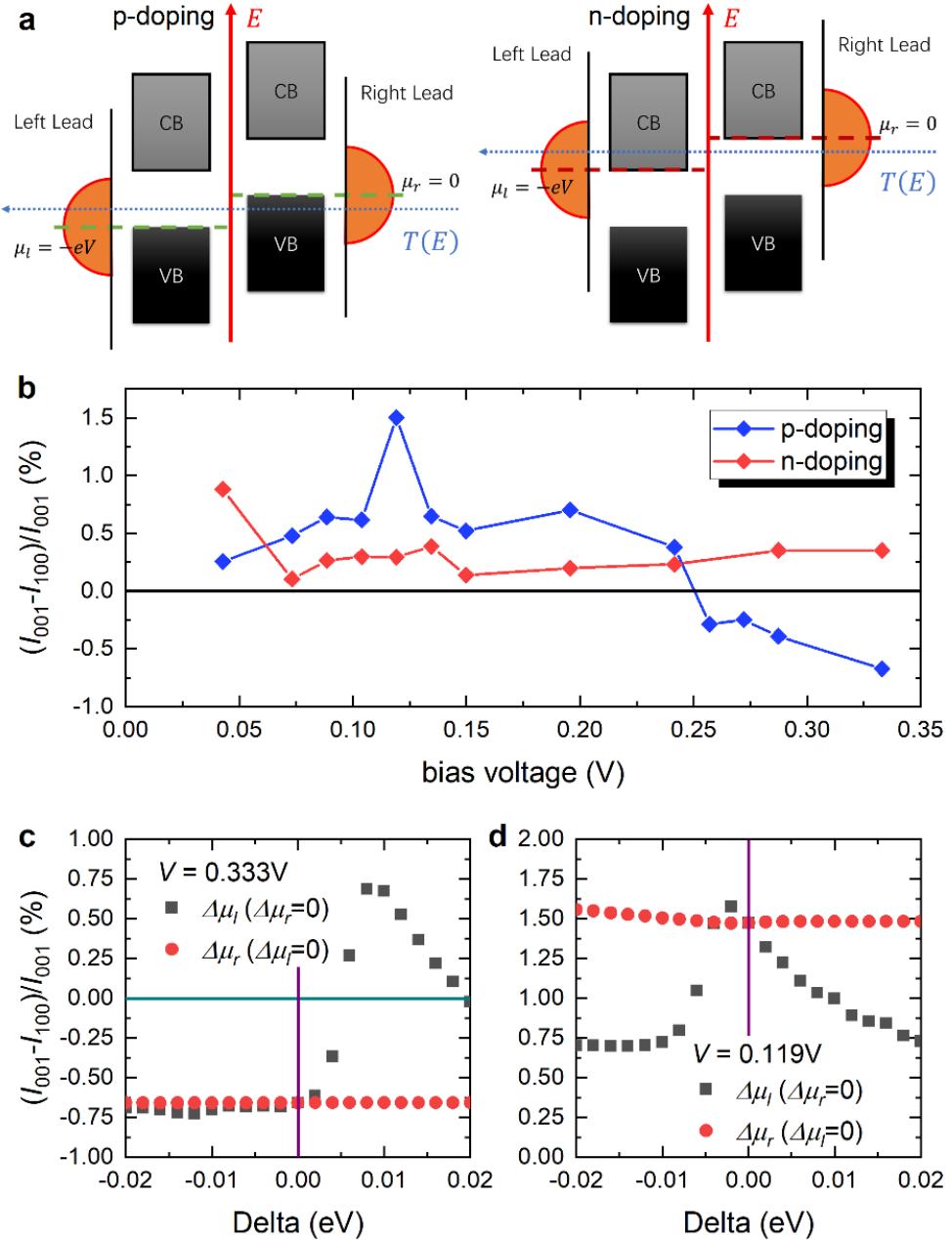
1 from **Figure 3a** is further detailed in **Figure 3b**. We see that the TAMR at $\theta_B = -90^\circ, 90^\circ$ and
 2 270° switches from negative to positive as the bias voltage becomes more negative. This bias-
 3 dependent sign change is similarly observed under lower background magnetic fields of 4.5 T
 4 (where the 4L CrPS₄ is in a canted spin state) and 1 T (where the 4L CrPS₄ is in the AFM state).
 5 In the 2L CrPS₄-based MTJ, a comparable bias voltage-dependent TAMR is also detected in
 6 the canted spin phases ($B = 5$ T and 1 T, as detailed in **Figure S5** in Supporting Information).
 7 Additionally, the gate-dependence of TAMR in the AFM state (at $B = 1$ T) is demonstrated in
 8 **Figure S6** (Supporting Information). These experimental findings collectively reinforce the
 9 notion that the electrically tunable nature of TAMR is a universal characteristic across few-
 10 layer CrPS₄ devices with different spin configurations. We note that these interesting features
 11 have not been reported thus far.

13 **2.3. First-principles Calculations to Understand the Electrically Tunable TAMR in CrPS₄**
 14 To gain an insight into the origin of electrically tunable and polarity reversal of the TAMR, we
 15 employ first-principles calculations to elucidate the tunneling transport in the OOP and IP
 16 directions of a bilayer CrPS₄-based tunneling junctions with top and bottom graphene electrodes.
 17 To simulate systems with bias voltages, in the calculations, we added an electric field ranging
 18 from 0.02 to 0.21 V/Å along the z-axis (OOP), corresponding to the interlayer bias voltage from
 19 0.043 V to 0.365 V in the 2L CrPS₄. Based on the non-equilibrium Green's function (NEGF)
 20 method^[63], the corresponding tunneling current I with bias voltage V is calculated along the
 21 OOP direction. The Wannier function (WF)-based tight-binding Hamiltonian of the 2L CrPS₄
 22 from first-principles calculations is used as the central part. To simplify the device structure,
 23 the single *s* orbital non-magnetic atomic chain is treated as the lead, which connects to the p_z
 24 orbitals of the top and bottom *S* atoms. No barrier between the leads and the 2L CrPS₄ is
 25 assumed. As shown in **Figure 4a**, with p-type doping, the chemical potentials for the left (μ_l)
 26 and right (μ_r) leads are set to the valence band maximum (VBM) of the upper and lower layers,
 27 respectively, while with n-type doping, those are set to the conduction band minimum (CBM)
 28 of both layers. The transmission rate $T(E)$ as a function of E , the kinetic energy of the injection
 29 electrons from leads, is obtained with a 96×96 k-points mesh in the Brillouin zone. So that
 30 the tunneling current I is given by

$$33 \quad I = -\frac{e}{h} \int_E^{\square} dE T(E) [f(E - \mu_l) - f(E - \mu_r)]$$

31 where e is the electron charge, h is Planck's constant, and $f(E)$ is the Fermi-Dirac distribution
 32 function with the temperature $k_B T = 1.0$ meV. Here, $V = -1/e (\mu_l - \mu_r)$ is the bias voltage.

1 $I - V$ relationship is obtained for two ferromagnetic spin states: one with (001) spin direction
 2 and the other with (100) spin direction, corresponding to the spin-polarized states under the
 3 OOP and IP magnetic fields of about 7 T, respectively. **Figure 4b** shows the difference rate of
 4 tunneling current, $\frac{I(001)-I(100)}{I(001)} \times 100\%$, equivalent to the tunneling anisotropic
 5 magnetoconductance, as a function of bias voltage for both n-type and p-type doping situations.
 6 In the p-type doping region, a transition from a positive to a negative value appears at about



7 **Figure 4.** First-principles calculation for tunneling current along OOP (001) and IP (001) directions in
 8 bilayer CrPS₄. a) The tunneling model of the bilayer CrPS₄ under bias V , left and right panels
 9 correspond to p-doping and n-doping, respectively. CB and VB correspond to conduction bands and
 10 valence bands, respectively. b) The difference rate of tunnelling current $\frac{[I(001)-I(100)]}{I(001)} \times 100\%$ as a
 11

1 function of bias voltage. With p-doping and $V = 0.333\text{V}$ (c) and 0.119V (d), the rate $\frac{[I(001)-I(100)]}{I(001)} \times$
 2 100% as a function of a small shift Δ of μ_l ($\Delta\mu_r = 0$) and μ_r ($\Delta\mu_l = 0$).

3
 4 $V = 0.25\text{V}$. In this model, positive and negative bias voltages have the symmetric $I - V$
 5 relation so that at about $V = -0.25\text{V}$, a similar sign reversal is expected, consistent with the
 6 experimental results (also see Supplementary Note 1). The resistance with OOP magnetization
 7 R_{\perp} is lower than that with in-plane magnetization R_{\parallel} , while $R_{\perp} > R_{\parallel}$ with $|V| > 0.25\text{V}$. The
 8 1% difference of the magnitude is consistent with the experimental observation between $\theta_B =$
 9 0° and 90° , and is mainly from the contribution of band shift of all bands between μ_l and μ_r
 10 due to anisotropic spin directions (See **Figure S7** in Supporting Information).

11 Both bias and gate voltages not only control the bias between two CrPS₄ layers but also
 12 potentially adjust the chemical potential for each layer. An energy offset should actually appear
 13 between the chemical potential and VBM (p-type doping)/CBM (n-type doping), leading to an
 14 asymmetric $I - V$ relationship. To simulate this asymmetric behavior, we made small shifts Δs
 15 of μ_l and μ_r , namely $\Delta\mu_l$ and $\Delta\mu_r$ respectively. Under the p-type doping environment, **Figure**
 16 **4c** and **4d** show the rate $\frac{I(001)-I(100)}{I(001)} \times 100\%$ as a function of $\Delta\mu_l$ and $\Delta\mu_r$ under $= 0.333\text{V}$ and
 17 0.119V , respectively. While only tiny change happens with finite $\Delta\mu_r$, positive and negative
 18 $\Delta\mu_l$ lead to distinct behaviors, which eventually cause the asymmetric $I - V$ curve.

19 Another possible mechanism behind the electrically tunable TAMR is involving a switch in the
 20 easy axis from OOP to IP orientation due to the applied bias or gate voltage^[64]. To investigate
 21 this possibility, we conducted a thorough analysis of our results under both bias and gate voltage
 22 conditions. In the case of gate voltage, as depicted in **Figures 2h** and **S6** for a 4L sample, we
 23 compared the changes in TAMR magnitude with varying gate voltages and magnetic fields (9
 24 T and 1 T). Notably, at $\theta_B = 90^\circ$, the TAMR magnitude shifts from 4% to -5% at 9 T and from
 25 3% to -3% at 1 T. These findings suggest that while doping-induced variation in the easy axis
 26 orientation cannot be discounted, their impact on the observed TAMR polarity switch appears
 27 limited. This assertion is supported by the fact that, if doping were the primary driver, a more
 28 significant effect would be expected at lower magnetic fields (1 T) compared to higher fields
 29 (9 T). Similarly, under bias conditions (**Figure 3**), we see a decrease in TAMR magnitude as
 30 the magnetic field reduces. These results suggest that while bias or gate voltage-induced
 31 changes in the easy axis orientation may contribute, they are not the dominant factor driving
 32 the observed TAMR polarity switching.

33 **3. Conclusion**

1 In conclusion, we have investigated the layer-dependent behavior of TMR in few-layer CrPS₄-
2 based MTJs. This investigation has not only illuminated the dependence of TMR on gate
3 voltage but has also uncovered the anomalous TMR oscillations under both OOP and IP
4 magnetic fields, potentially heralding the presence of a spin geometric phase. A cornerstone of
5 our study is the discovery of a highly tunable TAMR in the few-layer CrPS₄ through both bias
6 and gate voltages. This adaptability of TAMR, discernible across diverse states of CrPS₄,
7 including AFM, canted spin, and SP phases under various magnetic fields, marks a significant
8 advancement. Theoretical support for these experimental observations was provided through
9 first-principles calculations, which helped delineate the differences in electrical transport
10 properties between IP and OOP directions in a 2L CrPS₄ system. Our findings not only advance
11 the understanding of 2D magnetic materials but also uncover the novel quantum states, opening
12 new opportunities in the realm of materials science and spintronics.

13 **4. Methods**

14 **4.1. Synthesis of CrPS₄ crystals and device fabrication**

15 Single crystals of CrPS₄ were grown by chemical vapor transport using iodine as the transport
16 media. While under an atmosphere of pure argon, 0.52 g of metallic chromium (BTC 99.99%),
17 0.31 g of elemental phosphorus (Millipore-Sigma 99.99%), 1.30 g of sulfur powder (Alfa-Aesar
18 99.5%) and 0.025 g of iodine (Thermo-Fisher 99.99) were placed in a fused silica tube (0.9 cm
19 inner diameter × 20 cm length). The tube was then cooled to 77 K, evacuated to 50 mTorr, and
20 then sealed to an over-length of 15 cm. The tube contents were pre-reacted by heating at 25
21 K/min to 525 K, and maintaining that temperature for 20 hours after which they were cooled to
22 room temperature at 5 K/min. After vigorous shaking of the tube to re-mix the reagents, the
23 tube was placed in a two-zone furnace and heated at 5 K/min to 900 °C at the charge zone and
24 to 875 K at the growth zone. The tube was held at these conditions for 175 hours, after which
25 it was cooled at 25 K/min to room temperature. The tube was then opened under an argon
26 atmosphere, the crystals mechanically extracted, then sealed in glass scintillation vials for
27 further use.

28 The few-layer CrPS₄ tunneling junction was fabricated by a layer-by-layer dry transfer
29 method^[65,66]. Atomically thin CrPS₄, hBN and graphite were mechanically exfoliated from their
30 bulk crystals onto the SiO₂(200 nm)/Si substrates. The sample thickness was determined by an
31 atomic force microscope. For a monolayer CrPS₄, the thickness is of ~ 0.7 nm. The quality of
32 our exfoliated CrPS₄ flakes was further characterized by Raman spectroscopy (See **Figure S8**
33 in Supporting Information). The stack of hBN/graphite/CrPS₄/graphite/hBN was picked up one
34 by one using a polydimethylsiloxane stamp with a polyvinyl alcohol (PVA) layer on the top.

1 The entire stack was then released onto a SiO₂/Si substrate with prefabricated Pt/Ti (30nm/5nm)
2 electrodes, which were prepared by a standard nanofabrication procedure. After dissolving the
3 PVA layer in deionized water, a graphite flake was finally transferred onto the stack to serve as
4 a top gate. Here, the hBN flakes serve as protection and dielectric layers, and graphite flakes
5 serve as either bias or gate electrodes. To avoid any possible degradation of the thin CrPS₄
6 layers, the exfoliation and the transfer processes were performed in an argon-filled glove box
7 with H₂O and O₂ concentrations of < 0.1 ppm.

8 **4.2. Electrical and magnetotransport measurements**

9 The low temperature electrical and magnetotransport measurements were carried out inside a
10 closed cycle ⁴He cryostat (Oxford TeslatronPT with a base temperature of 1.5 K). The angle
11 between the sample and magnetic field was controlled by rotating the sample via a self-written
12 program. The DC electrical transport measurements were performed with a Keithley 2400
13 Sourcemeter. We note that all the results for 2L (or 3L and 4L) (or “for a specific layer) CrPS₄
14 tunneling device in this work were obtained from the same device.

15 **4.3. Raman spectroscopy measurements**

16 The Raman spectra were acquired through a Horiba confocal Raman microscope with a 532 nm
17 laser excitation. A 2400 grooves/mm grating was used to achieve a spectral resolution of below
18 1.4 cm⁻¹. The Raman spectra taken at different temperatures were conducted in a cooling stage
19 down to liquid nitrogen temperature with an optical window (INSTECH, INC, model HCP421V-
20 PMH).

21 **4.4. First-principles calculations**

22 Our first-principles calculations are performed to calculate electronic and magnetic properties
23 of bilayer CrPS₄ by using the projector-augmented wave pseudopotential ^[67] implemented in
24 the VASP package^[68,69]. Generalized gradient approximation in Perdew–Burke–Ernzerhofer
25 formation^[70] is employed as the exchange–correlation potential, and the Hubbard *U* method ^[71]
26 is introduced to treat localized 3d orbitals of Cr atoms, using *U* = 2.5 eV as previously tested.
27 An energy cutoff of 500 eV is used for the plane-wave expansion throughout the calculations.
28 The Γ -centered 2D k-points mesh of 5×7 is sampled in the Brillouin zone. A vacuum region of
29 15 Å is chosen to prevent artificial interactions between neighboring sheets along the z direction.
30 The interlayer interactions in the bilayer CrPS₄ are considered by adopting the DFT-D3 method
31 ^[72] to describe long-ranged van der Waals interactions. The structures are fully relaxed until the
32 force on each atom is smaller than 0.01 eV/Å, and the total energy convergence criterion is set
33 as 10⁻⁷ eV. Spin-orbit coupling is included in self-consistent electronic calculations.

1 A unitary transformation of Bloch waves was performed to construct the tight-binding
2 Hamiltonian in a WF basis implemented in the WANNIER90 package^[73]. A WF-based
3 Hamiltonian has exactly the same eigenvalues as those obtained by first-principles calculations
4 among all occupied bands and the bands below 0.5eV to CBM.

5

6 **Supporting Information**

7 Supporting Information is available from the Wiley Online Library or from the author.

8

9 **Acknowledgements**

10 J.T. also acknowledges the financial support of the U.S. National Science Foundation (NSF)
11 grant 2228841 for data analysis, H. H. and J. Y. acknowledge the financial support of the
12 National Natural Science Foundation of China (Grant Number 12274309), K.W. and T.T.
13 acknowledge support from the JSPS KAKENHI (Grant Numbers 20H00354 and 23H02052)
14 and World Premier International Research Center Initiative (WPI), MEXT, Japan.

15

16 **Competing Interests**

17 The authors declare no competing interests.

18

19 **References**

20 [1] C. Gong, L. Li, Z. Li, H. Ji, A. Stern, Y. Xia, T. Cao, W. Bao, C. Wang, Y. Wang, Z.
21 Q. Qiu, R. J. Cava, S. G. Louie, J. Xia, X. Zhang, *Nature* **2017**, *546*, 265.

22 [2] B. Huang, G. Clark, E. Navarro-Moratalla, D. R. Klein, R. Cheng, K. L. Seyler, Di.
23 Zhong, E. Schmidgall, M. A. McGuire, D. H. Cobden, W. Yao, D. Xiao, P. Jarillo-Herrero, X.
24 Xu, *Nature* **2017**, *546*, 270.

25 [3] J. Lee, T. Y. Ko, J. H. Kim, H. Bark, B. Kang, S. G. Jung, T. Park, Z. Lee, S. Ryu, C.
26 Lee, *ACS Nano* **2017**, *11*, 10935.

27 [4] M. Bonilla, S. Kolekar, Y. Ma, H. C. Diaz, V. Kalappattil, R. Das, T. Eggers, H. R.
28 Gutierrez, M. H. Phan, M. Batzill, *Nat. Nanotechnol.* **2018**, *13*, 289.

29 [5] Y. Deng, Y. Yu, Y. Song, J. Zhang, N. Z. Wang, Z. Sun, Y. Yi, Y. Z. Wu, S. Wu, J.
30 Zhu, J. Wang, X. H. Chen, Y. Zhang, *Nature* **2018** **2018**, *563*, 94.

31 [6] B. Huang, G. Clark, D. R. Klein, D. MacNeill, E. Navarro-Moratalla, K. L. Seyler, N.
32 Wilson, M. A. McGuire, D. H. Cobden, D. Xiao, W. Yao, P. Jarillo-Herrero, X. Xu, *Nat.*
33 *Nanotechnol.* **2018**, *13*, 544.

1 [7] S. Jiang, L. Li, Z. Wang, K. F. Mak, J. Shan, *Nat. Nanotechnol.* **2018**, *13*, 549.

2 [8] S. Jiang, J. Shan, K. F. Mak, *Nat. Mater.* **2018**, *17*, 406.

3 [9] Z. Wang, D. Sapkota, T. Taniguchi, K. Watanabe, D. Mandrus, A. F. Morpurgo, *Nano*
4 *Lett.* **2018**, *18*, 4303.

5 [10] L. Webster, J.-A. Yan, *Phys. Rev. B* **2018**, *98*, 144411.

6 [11] W. Chen, Z. Sun, Z. Wang, L. Gu, X. Xu, S. Wu, C. Gao, *Science* **2019**, *366*, 983.

7 [12] D. R. Klein, D. MacNeill, Q. Song, D. T. Larson, S. Fang, M. Xu, R. A. Ribeiro, P. C.
8 Canfield, E. Kaxiras, R. Comin, P. Jarillo-Herrero, *Nat. Phys.* **2019**, *15*, 1255.

9 [13] T. Song, Z. Fei, M. Yankowitz, Z. Lin, Q. Jiang, K. Hwangbo, Q. Zhang, B. Sun, T.
10 Taniguchi, K. Watanabe, M. A. McGuire, D. Graf, T. Cao, J. H. Chu, D. H. Cobden, C. R.
11 Dean, D. Xiao, X. Xu, *Nat. Mater.* **2019**, *18*, 1298.

12 [14] S. O. Valenzuela, S. Roche, *Nat. Nanotechnol.* **2019**, *14*, 1088.

13 [15] Y. Deng, Y. Yu, M. Z. Shi, Z. Guo, Z. Xu, J. Wang, X. H. Chen, Y. Zhang, *Science*
14 **2020**, *367*, 895.

15 [16] X. X. Zhang, L. Li, D. Weber, J. Goldberger, K. F. Mak, J. Shan, *Nat. Mater.* **2020**,
16 *19*, 838.

17 [17] R. Zhu, W. Zhang, W. Shen, P. K. J. Wong, Q. Wang, Q. Liang, Z. Tian, Y. Zhai, C.
18 W. Qiu, A. T. S. Wee, *Nano Lett.* **2020**, *20*, 5030.

19 [18] A. Bedoya-Pinto, J.-R. Ji, A. K. Pandeya, P. Gargiani, M. Valvidares, P. Sessi, J. M.
20 Taylor, F. Radu, K. Chang, S. S. P. Parkin, *Science* **2021**, *374*, 616.

21 [19] Y. Xu, A. Ray, Y. T. Shao, S. Jiang, K. Lee, D. Weber, J. E. Goldberger, K.
22 Watanabe, T. Taniguchi, D. A. Muller, K. F. Mak, J. Shan, *Nat. Nanotechnol.* **2022**, *17*, 143.

23 [20] C. Ye, C. Wang, Q. Wu, S. Liu, J. Zhou, G. Wang, A. Söll, Z. Sofer, M. Yue, X. Liu,
24 M. Tian, Q. Xiong, W. Ji, X. Renshaw Wang, *ACS Nano* **2022**, *16*, 11876.

25 [21] K. S. Burch, D. Mandrus, J. G. Park, *Nature* **2018**, *563*, 47.

26 [22] M. Gibertini, M. Koperski, A. F. Morpurgo, K. S. Novoselov, *Nat. Nanotechnol.* **2019**,
27 *14*, 408.

28 [23] H. Kurebayashi, J. H. Garcia, S. Khan, J. Sinova, S. Roche, *Nature Reviews Physics*
29 **2022**, *4*, 150.

30 [24] M. Phan. New Research Trends in Electrically Tunable 2D van der Waals Magnetic
31 Materials. (Preprint) arXiv:2303.05748, submitted: Mar 2023

32 [25] K. F. Mak, J. Shan, D. C. Ralph, *Nature Reviews Physics* **2019**, *1*, 646.

33 [26] L. Thiel, Z. Wang, M. A. Tschudin, D. Rohner, I. Gutiérrez-Lezama, N. Ubrig, M.
34 Gibertini, E. Giannini, A. F. Morpurgo, P. Maletinsky, *Science* **2019**, *364*, 973.

1 [27] W. Jin, H. H. Kim, Z. Ye, S. Li, P. Rezaie, F. Diaz, S. Siddiq, E. Wauer, B. Yang, C.
2 Li, S. Tian, K. Sun, H. Lei, A. W. Tsen, L. Zhao, R. He, *Nat. Commun.* **2018**, *9*, 1.

3 [28] Z. Ni, A. V. Haglund, H. Wang, B. Xu, C. Bernhard, D. G. Mandrus, X. Qian, E. J.
4 Mele, C. L. Kane, L. Wu, *Nat. Nanotechnol.* **2021**, *16*, 782.

5 [29] Z. Lin, M. Lohmann, Z. A. Ali, C. Tang, J. Li, W. Xing, J. Zhong, S. Jia, W. Han, S.
6 Coh, W. Beyermann, J. Shi, *Phys. Rev. Mater.* **2018**, *2*, 1.

7 [30] P. Zhang, T. Chung, Q. Li, S. Wang, Q. Wang, W. L. B. Huey, S. Yang, J. E.
8 Goldberger, J. Yao, X. Zhang, *Nat. Mater.* **2022**, *21*, 1373.

9 [31] D. R. Klein, D. MacNeill, J. L. Lado, D. Soriano, E. Navarro-Moratalla, K. Watanabe,
10 T. Taniguchi, S. Manni, P. Canfield, J. Fernández-Rossier, P. Jarillo-Herrero, *Science* **2018**,
11 *360*, 1218.

12 [32] T. Song, X. Cai, M. W. Y. Tu, X. Zhang, B. Huang, N. P. Wilson, K. L. Seyler, L.
13 Zhu, T. Taniguchi, K. Watanabe, M. A. McGuire, D. H. Cobden, D. Xiao, W. Yao, X. Xu,
14 *Science* **2018**, *360*, 1214.

15 [33] S. Jiang, L. Li, Z. Wang, J. Shan, K. F. Mak, *Nat. Electron.* **2019**, *2*, 159.

16 [34] H. H. Kim, B. Yang, T. Patel, F. Sfigakis, C. Li, S. Tian, H. Lei, A. W. Tsen, *Nano*
17 *Lett.* **2018**, *18*, 4885.

18 [35] T. Song, M. W. Y. Tu, C. Carnahan, X. Cai, T. Taniguchi, K. Watanabe, M. A.
19 McGuire, D. H. Cobden, D. Xiao, W. Yao, X. Xu, *Nano Lett.* **2019**, *19*, 915.

20 [36] Z. E. Fu, P. I. Samarawickrama, Y. Zhu, Z. Mao, W. Wang, K. Watanabe, T.
21 Taniguchi, J. Tang, J. Ackerman, J. Tian, *Nano Lett.* **2023**, *27*, 58.

22 [37] Z. E. Fu, P. I. Samarawickrama, J. Ackerman, Y. Zhu, Z. Mao, K. Watanabe, T.
23 Taniguchi, W. Wang, Y. Dahnovsky, M. Wu, T. Chien, J. Tang, A. H. MacDonald, H. Chen,
24 J. Tian, *Nat. Commun.* **2024**, *15*, 3630.

25 [38] H. H. Kim, B. Yang, S. Li, S. Jiang, C. Jin, Z. Tao, G. Nichols, F. Sfigakis, S. Zhong,
26 C. Li, S. Tian, D. G. Cory, G. X. Miao, J. Shan, K. F. Mak, H. Lei, K. Sun, L. Zhao, A. W.
27 Tsen, *Proc. Natl. Acad. Sci. USA* **2019**, *116*, 11131.

28 [39] Z. Wang, I. Gutiérrez-Lezama, N. Ubrig, M. Kroner, M. Gibertini, T. Taniguchi, K.
29 Watanabe, A. Imamoğlu, E. Giannini, A. F. Morpurgo, *Nat. Commun.* **2018**, *9*, 2516.

30 [40] D. Ghazaryan, M. T. Greenaway, Z. Wang, V. H. Guarochico-Moreira, I. J. Vera-
31 Marun, J. Yin, Y. Liao, S. V. Morozov, O. Kristanovski, A. I. Lichtenstein, M. I. Katsnelson,
32 F. Withers, A. Mishchenko, L. Eaves, A. K. Geim, K. S. Novoselov, A. Misra, *Nat. Electron.*
33 **2018**, *1*, 344.

1 [41] H. H. Kim, S. Jiang, B. Yang, S. Zhong, S. Tian, C. Li, H. Lei, J. Shan, K. F. Mak, A.
2 W. Tsen, *Advanced Materials* **2020**, *32*, 1.

3 [42] M. J. A. Houmes, S. Mañas-Valero, A. Bermejillo-Seco, E. Coronado, P. G.
4 Steeneken, H. S. J. van der Zant, *Adv. Funct. Mater.* **2023**, *34*, 2310206.

5 [43] F. Wu, M. Gibertini, K. Watanabe, T. Taniguchi, I. Gutiérrez-Lezama, N. Ubrig, A. F.
6 Morpurgo, *Advanced Materials* **2023**, *35*, 2211653.

7 [44] F. Wu, M. Gibertini, K. Watanabe, T. Taniguchi, I. Gutiérrez-Lezama, N. Ubrig, A. F.
8 Morpurgo, *Nano Lett.* **2023**, *23*, 8140.

9 [45] M. Huang, J. C. Green, J. Zhou, V. Williams, S. Li, H. Lu, D. Djugba, H. Wang, B.
10 Flebus, N. Ni, C. R. Du, *Nano Lett.* **2023**, *23*, 8099.

11 [46] Y. Peng, Z. Lin, G. Tian, J. Yang, P. Zhang, F. Wang, P. Gu, X. Liu, C. W. Wang, M.
12 Avdeev, F. Liu, D. Zhou, R. Han, P. Shen, W. Yang, S. Liu, Y. Ye, J. Yang, *Adv. Funct.*
13 *Mater.* **2022**, *32*, 2106592.

14 [47] J. Yang, S. Fang, Y. Peng, S. Liu, B. Wu, R. Quhe, S. Ding, C. Yang, J. Ma, B. Shi, L.
15 Xu, X. Sun, G. Tian, C. Wang, J. Shi, J. Lu, J. Yang, *Phys. Rev. Appl.* **2021**, *16*, 024011.

16 [48] J. Son, S. Son, P. Park, M. Kim, Z. Tao, J. Oh, T. Lee, S. Lee, J. Kim, K. Zhang, K.
17 Cho, T. Kamiyama, J. H. Lee, K. F. Mak, J. Shan, M. Kim, J. G. Park, J. Lee, *ACS Nano*
18 **2021**, *15*, 16904.

19 [49] S. N. Neal, K. R. O'Neal, A. V. Haglund, D. G. Mandrus, H. A. Bechtel, G. L. Carr,
20 K. Haule, D. Vanderbilt, H. S. Kim, J. L. Musfeldt, *2D Mater.* **2021**, *8*, 035020.

21 [50] Y. Peng, S. Ding, M. Cheng, Q. Hu, J. Yang, F. Wang, M. Xue, Z. Liu, Z. Lin, M.
22 Avdeev, Y. Hou, W. Yang, Y. Zheng, J. Yang, *Advanced Materials* **2020**, *32*, 2001200.

23 [51] M. Joe, H. Lee, M. M. Alyörük, J. Lee, S. Y. Kim, C. Lee, J. H. Lee, *Journal of*
24 *Physics Condensed Matter* **2017**, *29*, 405801.

25 [52] Q. L. Pei, X. Luo, G. T. Lin, J. Y. Song, L. Hu, Y. M. Zou, L. Yu, W. Tong, W. H.
26 Song, W. J. Lu, Y. P. Sun, *J. Appl. Phys.* **2016**, *119*, 043902.

27 [53] D. K. Wal, A. Iwens, T. Liu, P. Tang, G. E. W. Bauer, and B. J. Wees, *Phys. Rev.*
28 *B* **2023**, *107*, 180403.

29 [54] R. Wu, A. Ross, S. Ding, Y. Peng, F. He, Y. Ren, R. Lebrun, Y. Wu, Z. Wang, J.
30 Yang, A. Brataas, and M. Kläui, *Phys. Rev. Applied* **2022**, *17*, 064038.

31 [55] S. L. Bud'Ko, E. Gati, T. J. Slade, P. C. Canfield, *Phys. Rev. B* **2021**, *103*, 224407.

32 [56] S. Calder, A. V Haglund, Y. Liu, D. M. Pajerowski, H. B. Cao, T. J. Williams, V. O.
33 Garlea, D. Mandrus, *Phys. Rev. B* **2020**, *102*, 024408.

34 [57] A. Louisy, G. Ouvrard, D. M. Schleich, R. Brec, *Solid State Commun.* **1978**, *28*, 61.

1 [58] D. Loss, P. Goldbart, A. V. Balatsky, *Phys. Rev. Lett.* **1990**, *65*, 1655.

2 [59] D. Loss, P. M. Goldbart, *Phys. Rev. B* **1992**, *45*, 15.

3 [60] A. G. Aronov, Y. B. Lyanda-Geller, *Phys. Rev. Lett.* **1993**, *70*, 343.

4 [61] T. Z. Qian, Z. Bin Su, *Phys. Rev. Lett.* **1994**, *72*, 2311.

5 [62] L. Gao, X. Jiang, S. H. Yang, J. D. Burton, E. Y. Tsymbal, S. S. P. Parkin, *Phys. Rev. Lett.* **2007**, *99*, 226602.

7 [63] S. Datta, *Quantum Transport : Atom to Transistor*, Cambridge University Press, Cambridge University Press, Cambridge, **2005**.

9 [64] I. A. Verzhbitskiy, H. Kurebayashi, H. Cheng. *et al. Nat. Electron.* **2020**, *3*, 460.

10 [65] Z. Fu, J. W. Hill, B. Parkinson, C. M. Hill, J. Tian, *2D Mater.* **2022**, *2*, 015022.

11 [66] P. Samarawickrama, R. Dulal, Z. Fu, U. Erugu, W. Wang, J. Ackerman, B. Leonard, J. Tang, T. Y. Chien, J. Tian, *ACS Omega* **2021**, *6*, 2966.

13 [67] P. E. Blöchl, *Phys. Rev. B* **1994**, *50*, 17953.

14 [68] G. Kresse, J. Furthmüller, *Comput. Mater. Sci.* **1996**, *6*, 15.

15 [69] R. A. Vargas-Hernández, *Journal of Physical Chemistry A* **2020**, *124*, 4053.

16 [70] J. P. Perdew, K. Burke, M. Ernzerhof, *Phys. Rev. Lett.* **1996**, *77*, 3865.

17 [71] S. Dudarev, G. Botton, *Phys. Rev. B* **1998**, *57*, 1505.

18 [72] S. Grimme, J. Antony, S. Ehrlich, H. Krieg, *Journal of Chemical Physics* **2010**, *132*, 154104.

20 [73] A. A. Mostofi, J. R. Yates, G. Pizzi, Y. S. Lee, I. Souza, D. Vanderbilt, N. Marzari, *Comput. Phys. Commun.* **2014**, *185*, 2309.