

# In situ investigation of the atomic structure of carbonate-silicate liquids at high pressure-temperature and spectroscopic characterization of the recovered quenched glasses



Veronica Stopponi <sup>a,b,\*</sup>, Annalisa D'Arco <sup>c</sup>, Yoshio Kono <sup>d</sup>, Federica Piccirilli <sup>b</sup>, Brent T. Poe <sup>e</sup>, Stefano Lupi <sup>c</sup>, Manuela Nazzari <sup>f</sup>, Lucia Pappalardo <sup>g</sup>, Giulia Marras <sup>a</sup>, Michele Zacchigna <sup>b</sup>, Craig E. Manning <sup>h</sup>, Claudia Romano <sup>i</sup>, Vincenzo Stagno <sup>a, \*\*</sup>

<sup>a</sup> Department of Earth Sciences, Sapienza University of Rome, Italy

<sup>b</sup> CNR - Istituto Officina dei Materiali (IOM), Trieste, Italy

<sup>c</sup> Department of Physics, Sapienza University of Rome, Italy

<sup>d</sup> Department of Physics and Astronomy, Kwansei Gakuin University, Sanda, Japan

<sup>e</sup> Department of Engineering and Geology, University of Chieti-Pescara, Italy

<sup>f</sup> National Institute of Geophysics and Volcanology, Rome, Italy

<sup>g</sup> National Institute of Geophysics and Volcanology, Osservatorio Vesuviano, Italy

<sup>h</sup> Department of Earth, Planetary and Space Sciences, University of California, Los Angeles, CA, USA

<sup>i</sup> Department of Sciences, University of Studies Roma Tre, Rome, Italy

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## ABSTRACT

Carbonate-silicate melts that originate in Earth's interior are described as transitional melts which possess compositions intermediate between carbonatitic and basaltic end members. The covariation of key oxides between carbonatite and basalt (e.g., 10–35 wt% SiO<sub>2</sub> and 40–10 wt% CO<sub>2</sub>, respectively) is expected to have a strong effect on liquid properties. However, due to their paucity both in the record of terrestrial rocks and as quenched glasses, their molecular structure has remained poorly explored to date. We investigated the atomic structure of a synthetic carbonate-silicate liquid with chemical composition within the CaO-MgO-Al<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub>-FeO-Na<sub>2</sub>O-CLO<sup>-</sup>-CO<sub>2</sub> oxide system having 18.28 wt% SiO<sub>2</sub> and 22.54 wt% CO<sub>2</sub> using multi-angle energy dispersive X-ray diffraction at pressures (P) and temperatures (T) of 1.4 GPa/1815 °C, 2.6 GPa/1865 °C, 4.3 GPa/1990 °C, 4.4 GPa/1950 °C. The results show that the intermediate range ordering of the structure decreases with an increase of both P and T. Based on this study, the carbonate-silicate magmas at upper mantle P-T conditions are expected to increase their viscosities during their ascent through the mantle as a result of increasing intermediate range ordering upon cooling and decompression. Additionally, spectroscopic measurements were carried out on the quenched glasses at ambient pressure using micro-Raman as well as micro-FTIR in reflection and transmission modes in the mid infrared range. High pressure investigation using micro-FTIR was also conducted. The distribution of Q<sub>n</sub> species obtained by deconvolution of the Raman spectra within the aluminosilicate region confirms the depolymerized nature of the quenched glasses as inferred by the low viscosities of the corresponding liquids; peculiar characteristics of the C vibrations would suggest a distorted environment surrounding the network modifying CO<sub>3</sub><sup>2-</sup> anion. No evidence of molecular CO<sub>2</sub> was detected. Notably, we find evidence of both dissolved molecular CO and CO linked to a metal cation forming carbonyl complexes in the quenched glasses at P-T-fo<sub>2</sub> conditions compatible with a hot Archean upper mantle. This suggests a role for carbonate-silicate magmas as carriers of reduced gaseous C-O-H species towards the early atmosphere along with the mobilization of PGE-elements.

\* Corresponding author at: CNR - Istituto Officina dei Materiali (IOM), Trieste, Italy

\*\* Corresponding author.

E-mail addresses: [stopponi@iom.cnr.it](mailto:stopponi@iom.cnr.it) (V. Stopponi), [vincenzo.stagno@uniroma1.it](mailto:vincenzo.stagno@uniroma1.it) (V. Stagno).

## 1. Introduction

Natural magmas are the most efficient carriers of  $\text{CO}_2$  from the interior of Earth to the surface and, eventually, out to the atmosphere. Extensive experimental studies were carried out in the last decades to determine phase equilibria and melting conditions at high pressure (P) and temperature (T) of mantle rocks (i.e. peridotites and eclogites) in the presence of oxidized carbon ( $\text{C}^{4+}$ ) (Wyllie and Huang, 1976; Eggler, 1978; Wallace and Green, 1988; Dalton and Wood, 1993; Dalton and Presnall, 1998; Moore and Wood, 1998; Hammouda, 2003; Yaxley and Brey, 2004; Gudfinnsson and Presnall, 2005; Dasgupta and Hirschmann, 2006, 2007; Ghosh et al., 2009, 2014; Litasov and Ohtani, 2009a, 2009b, 2010; Keshav and Gudfinnsson, 2010; Kiseeva et al., 2012, 2013; Green, 2015). These studies show that melts generated at near-solidus conditions are carbonatitic in composition (5–10 wt%  $\text{SiO}_2$ –40 wt%  $\text{CO}_2$ ) and, by increasing T, progressively dissolve more  $\text{SiO}_2$  (10–30 wt %) at the expense of  $\text{CO}_2$  (decreasing to ~15 wt%), while the melt fractions modelled on the basis of carbon available in the mantle (Dasgupta et al., 2013; Aiuppa et al., 2021) range from 0.01 wt% to ~1 wt% with transitional melts being stable over relatively large P-T conditions as a function of volatile components (i.e.,  $\text{CO}_2$  and  $\text{H}_2\text{O}$ ). As a result of the combined effect of P, T and bulk rock chemistry, a wide range of melt compositions exists spanning from carbonatite to kimberlite to basalt (Hammouda and Keshav, 2015). The rheological properties of melts in this compositional range likely vary significantly with changing concentrations of network forming vs. modifying cations and the mechanism of  $\text{CO}_2$  incorporation in the liquid structure (i.e., dissolved  $\text{CO}_3^{2-}$  and molecular  $\text{CO}_2$ ). Recently, the viscosity of melts intermediate between carbonate and silicate end members (here termed “transitional”) was shown to be about one order of magnitude higher than dolomitic melts at mantle P-T conditions, providing the first experimental evidence of the effect of increasing the amount of tetrahedrally-coordinated Si in ultrabasic magmas on viscosity, mobility and migration rate (Stagno et al., 2020a). Recent implementations that allow experimental measurements of the liquid structure to be performed by X-ray diffraction technique at HP-T (Kono et al., 2014a) provide the opportunity to link changes in viscosity to atomic-scale changes and distortion of the cation-coordinated polyhedra. However, to date, melt structure measurements conducted at high P-T applicable to natural magmatic systems are available only for dolomitic and calcitic liquids representative of carbonatitic melts (Kono et al., 2014b; Hudspeth et al., 2018), melilititic liquids (Stagno et al., 2020b) and liquids representative of basalts erupted at mid-ocean ridges (Sakamaki et al., 2013). No measurements have been conducted *in situ* on transitional liquids, although their presence at mantle conditions has been established through geochemical (Tappe et al., 2017) and experimental evidence (see Fig. 1 and references in Stagno et al., 2020a).

The P-T dependence of  $\text{CO}_2$  incorporation in the melt structure has been mostly investigated through the analyses of the quenched glasses providing, therefore, a valuable information on the solubility of  $\text{CO}_2$  for a specific composition. This can be summarized with a gradual increase of  $\text{CO}_2$  being dissolved as the melt is less polymerized (see Fig. 2 in Brooker et al., 2011). The effect of P on the  $\text{CO}_2$  solubility appears to be gradual for all compositions except in the case of carbonatitic and kimberlitic glasses. While carbonatitic glasses show no P dependence implying that carbonatite rocks at the surface might retain the  $\text{CO}_2$  content from the mantle source, kimberlitic melts release most of the dissolved  $\text{CO}_2$  below 0.5 GPa, as experimentally observed (Brooker et al., 2011). Importantly, recent data on  $\text{CO}_2$  solubility in basaltic and andesitic melts showed that the increase of  $\text{CO}_2$  with increasing P is more pronounced than what would be expected from Henry's law (Amalberti et al., 2021), in agreement with molecular dynamics simulations (Guillot and Sator, 2011). In contrast, a negative correlation was established between T and  $\text{CO}_2$  solubility in glasses regardless of their composition, with the exception of nephelinitic glasses which instead follow a positive trend (Blank and Brooker, 1994; Brooker et al., 1999).

In addition,  $\text{CO}_2$  solubility in glasses with a relatively low  $\text{SiO}_2$  content varying from ~30 to 45 wt% was observed to increase for decreasing  $\text{MgO}/(\text{MgO} + \text{CaO})$  content (Brooker et al., 2001a; Morizet et al., 2017a).

In contrast, the experimental investigation of the  $\text{CO}_2$  solubility in transitional melts has been quite problematic due to the difficulty in quenching these liquids as glasses. Instead, solubilities have been extrapolated from partially crystallized experiments (Brey and Ryabchikov, 1994; Brooker et al., 2011) and atomistic simulations (Guillot and Sator, 2011; Vuilleumier et al., 2015). Moussallam et al. (2015) published data on  $\text{CO}_2$  solubility as a function of the P and T of glass synthesis for compositions varying between 18 and 25 wt%  $\text{SiO}_2$ . Interestingly, a sudden decrease of  $\text{CO}_2$  solubility was observed at about 0.1 GPa that would imply a dramatic release of about 10 wt%  $\text{CO}_2$  upon decompression. The knowledge of the rheological properties of carbonate-silicate melts and their role as carriers of deep C requires the determination of the  $\text{CO}_2$  speciation in these liquids in response to the introduction and increase of tetrahedrally coordinated  $\text{Si}^{4+}$  cations in the evolving melt structure.

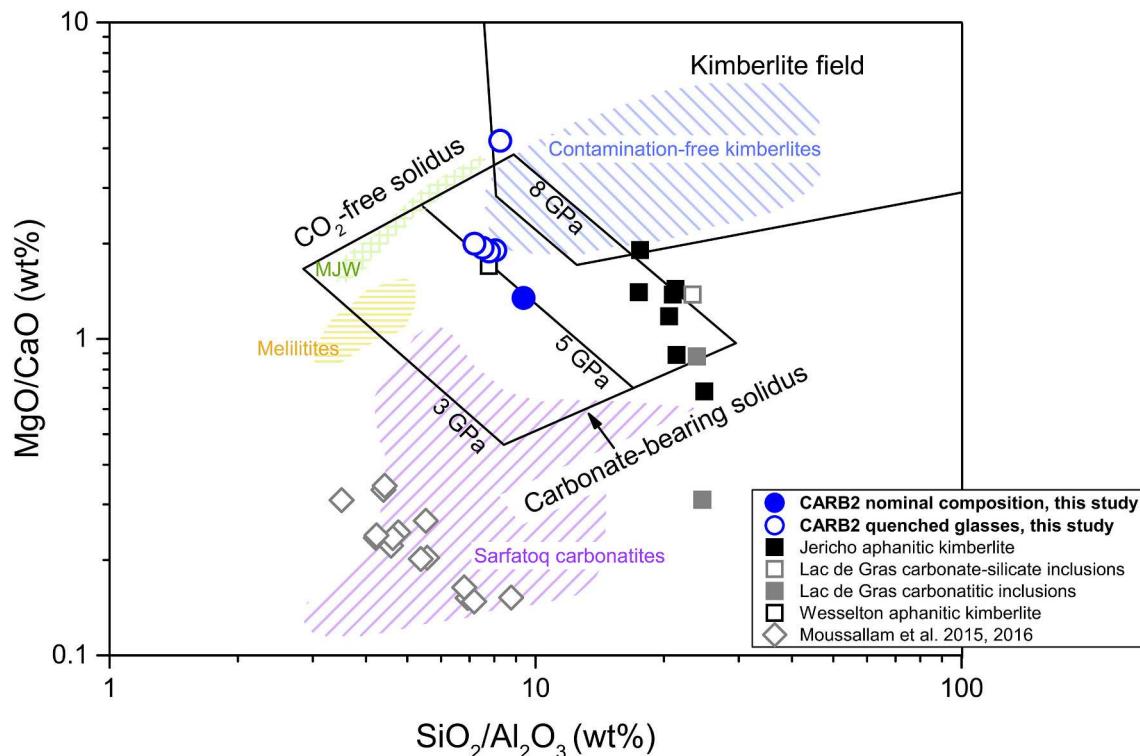
Pioneering experimental studies performed on quenched glasses at room/high P and T using both Fourier-transform infrared (FTIR) and Raman spectroscopy showed that C can dissolve in silicate liquids either as  $\text{CO}_2$  molecules or carbonate groups  $\text{CO}_3^{2-}$  (Brey and Green, 1976; Mysen et al., 1975). Further studies have investigated the effect of P, T and glass composition on the speciation of C within a wide spectrum of natural and synthetic glass compositions (Blank and Brooker, 1994; Mysen and Richet, 2019; Ni and Keppler, 2013). Currently available data suggest that  $\text{CO}_2$  dissolves predominantly as carbonate in (ultra) basic glasses (e.g. nephelinites), but as molecular  $\text{CO}_2$  in more acidic glasses (e.g. rhyolites, Blank and Brooker, 1994), with the fraction of molecular  $\text{CO}_2$  expressed as  $\text{CO}_2/(\text{CO}_2 + \text{CO}_3^{2-})$  that increases from 0.1 to 0.9 for  $\text{SiO}_2$  contents ranging from 65 to 85 wt%, at least at low T (Brooker et al., 1999; King and Holloway, 2002; Guillot and Sator, 2011; Konschak and Keppler, 2014).

*In situ* spectroscopic measurements on synthetic  $\text{CO}_2$ -bearing glasses at HP-T are very rare and only limited to dacite, phonolite, basaltic andesite, and alkali silicate melt at P of 1 atm–11.4 GPa at ambient T to 1000 °C (Konschak and Keppler, 2014).

These experiments are unique in showing that 1) as P increases the dissolved molecular  $\text{CO}_2$  is converted into carbonate anion with no distinction between dacitic and phonolitic glasses; 2) alkali silicate and basaltic andesite glasses show no evidence of molecular  $\text{CO}_2$  even when heated, contrary to that which is observed for phonolitic and dacitic glasses. Following these results, exsolution of molecular  $\text{CO}_2$  from the magma upon decompression is expected to be triggered by free (i.e., not-network forming) carbonate in the case of less polymerized magmas. A similar behavior is expected in the case of more basic-to-ultrabasic compositions such as carbonate-silicate magmas.

Transmission FTIR and Raman spectra of  $\text{CO}_2$ -rich silicate glasses by Moussallam et al. (2016) showed evidence of the occurrence of dissolved  $\text{CO}_2$  only in the form of carbonate instead of molecular  $\text{CO}_2$ . These glasses, however, are unrepresentative of realistic carbonate-silicate glasses because they are more Ca-rich and Mg-poor than partial melts of carbonated mantle peridotites and eclogites (Fig. 1). Fig. 1 illustrates that glasses investigated by Moussallam et al. (2015, 2016) fall into the field of carbonatites, while compositions explored in this study fall into the wide field of transitional melts in between carbonatites and kimberlites.

The different chemical composition of the transitional melts might result in different spectroscopic properties and  $\text{CO}_2$  speciation. Here, we present results relative to the atomic structure of carbonate-silicate transitional melts investigated by multi-angle energy-dispersive X-ray diffraction at HP-T and the characterization of the recovered quenched glasses by micro-Raman spectroscopy at ambient P and micro-FTIR spectroscopy both at ambient and high P. Diffraction data collected on the melt were, then, interpreted in light of recent viscosity



**Fig. 1.**  $\text{SiO}_2/\text{Al}_2\text{O}_3$  versus  $\text{MgO}/\text{CaO}$  discrimination diagram (Rock, 1991) modified after Gudfinnsson and Presnall (2005) showing both CARB2 starting material and the recovered synthetic glasses from the HP-T experiments in this study compared with natural  $\text{CO}_2$ -rich volcanic rocks (kimberlite field: Mitchell, 1986; melilitite field: Mitchell, 1996; Jericho: Price et al., 2000; Wesselton: Shee, 1986; Lac de Gras carbonatitic inclusions: Van Achterbergh et al., 2002; Lac de Gras silico-carbonate: Van Achterbergh et al., 2004) along with glasses investigated by Moussallam et al. (2015, 2016). The field MJW stands for the compositions of melts generated in equilibrium with the garnet lherzolite phase assemblage at 3–7 GPa in experiments by Walter (1998).

measurements for these compositions, while spectra from Raman and FTIR analyses were compared with data from literature to establish how the HP-T of synthesis can affect the  $\text{CO}_2$  solubility, carbon speciation and silicate network polymerization.

## 2. Experimental methods and analytical techniques

### 2.1. Synthetic starting material

The starting sample used in this study is a partially crystallized glass prepared by melting a mixture of oxides and carbonates at  $\sim 0.3$  GPa and  $1410$  ( $\pm 10$ )  $^{\circ}\text{C}$  for 1 h using a non-end loaded piston cylinder (details are in Stagno et al., 2020a). The starting material (CARB2) has the following composition: 18.28 wt%  $\text{SiO}_2$ , 20.43 wt%  $\text{CaO}$ , 27.50 wt%  $\text{MgO}$ , 6.72 wt%  $\text{FeO}$ , 1.95 wt%  $\text{Al}_2\text{O}_3$ , 1.37 wt%  $\text{Na}_2\text{O}$ , 1.21 wt%  $\text{ClO}^-$ , and 22.54 wt%  $\text{CO}_2$ . This chemical composition is representative of the quenched carbonate-silicate melts reported from several experimental studies (Dalton and Presnall, 1998; Moore and Wood, 1998; Gudfinnsson and Presnall, 2005; Brey et al., 2008; Foley et al., 2009; Litasov and Ohtani, 2009a, 2009b; Litasov and Ohtani, 2010; Stagno and Frost, 2010; Brey et al., 2011; Brooker and Kjarsgaard, 2011; Rohrbach and Schmidt, 2011; Kiseeva et al., 2012, 2013; Ghosh et al., 2014; Keshav and Gudfinnsson, 2014). The starting material CARB2 was used for the viscosity measurements reported in Stagno et al. (2020a), followed by the melt structure measurements described here.

### 2.2. In situ liquid structure measurements

Measurements of the liquid structure were performed at high P-T by in situ multi-angle energy dispersive synchrotron X-ray diffraction (EDXD) technique using the Paris-Edinburgh press at beamline 16BM-B of the Advanced Photon Source at Argonne National Laboratory (Lem-

ont, IL, USA; Kono et al., 2014a). The experimental P was determined using the equation of state of  $\text{MgO}$  collected from the  $\text{MgO}$  sleeve (Kono et al., 2010), while the T was raised following the power vs. temperature calibration (Kono et al., 2014a). For each run, the starting material was placed into a graphite capsule and enclosed in the cell assembly described by Stagno et al. (2020a). In some runs an additional layer of  $\text{CaCO}_3$  about 100  $\mu\text{m}$  thick (run CARB2\_1\_2016) or San Carlos olivine +  $\text{Li}_2\text{CO}_3$  of about 100  $\mu\text{m}$  thick (run CARB2\_2\_2018) was placed in the top-most part of the capsule in order to conduct two viscosity measurements at two different temperatures (i.e. the melting T of CARB2 starting material and melting T of either carbonates or olivine at given pressure) within the same experimental run (Stagno et al., 2020a). Liquid structure measurements were performed just after the viscosity experiments (Stagno et al., 2020a). Series of EDXD patterns were acquired within 3–4 h. The diffraction patterns were collected at fixed  $2\theta$  angles of  $3^\circ$  (625 s),  $4^\circ$  (550 s),  $5^\circ$  (500 s),  $7^\circ$  (700 s),  $9^\circ$  (500 s),  $12^\circ$  (500 s),  $16^\circ$  (500 s),  $22^\circ$  (625 s),  $28^\circ$  (1250 s) with a Ge-solid state detector mounted on a  $2\theta$  arm on a Huber rotation stage. The X-ray beam was adjusted to the center of the sample using the Prosilica GC1380 high-resolution camera with pixel size of 0.945  $\mu\text{m}/\text{pixel}$ . The slit sizes of the incident and diffracted X-rays were adjusted at every  $2\theta$  angle, to fix the Ge-SSD dead time at less than 15%. The EDXD patterns were analyzed using the software package provided by the beamline (Kono et al., 2014a) to obtain the structure factor  $S(q)$  and the reduced pair distribution function  $G(r)$ , respectively.  $S(q)$  gives important information about the intermediate range ordering of the melt structure and through its Fourier transform  $G(r)$ , distances between atom pairs might be determined. After EDXD acquisition was completed experiments were quenched by shutting down the power, then decompressed.

### 2.3. Chemical analyses and textural observations of the recovered quenched products

The recovered Paris-Edinburgh (PE) cells from the quenched runs were opened and the extracted capsules were embedded in epoxy resin and polished using ethanol to perform analyses to assess their chemical composition, textural features and, finally, spectroscopic properties. The recovered experimental products, all contained in a graphite capsule, were first observed under an optical microscope and showed the homogeneous surface of the quenched glasses (Fig. S1a-e of the Supplementary Materials). Textural observations were performed on carbon coated polished samples using the FEI QUANTA 400 scanning electron microscope available at the Department of Earth Sciences of Sapienza University (Rome). The chemical composition of the quenched products was analyzed using the electron microprobe (EMPA) Jeol JXA-8200 Superprobe available at Istituto Nazionale di Geofisica e Vulcanologia (INGV, Rome). An accelerating voltage of 15 kV, beam current of 7.5 nA and defocused beam with a diameter of 5  $\mu\text{m}$  were employed as operating conditions. For an accurate chemical composition, the following standards were used: Albite (Na, Si, Al), Apatite (Ca), Forsterite (Mg), Tugtupite (Cl) and Pyrite (Fe). Given the peculiarity of our quenched glasses, EMPA and SEM analyses were additionally carried out for comparison at the Bayerisches GeoInstitut (BGI, Germany) using JEOL JXA-8200 and Zeiss Gemini 1530, and repeated at INGV only on samples CARB2\_2\_2016, CARB2\_4 and CARB2\_2\_2018.

### 2.4. Micro-Raman spectroscopy

Raman spectra were collected on the polished samples after removal of the C coating using the Horiba LabRam HR 800 spectrometer available at the Department of Science of University of RomaTre (Italy). Data were collected with a 600 grooves/mm spectrometer grating and a CCD detector. A Nd-YAG laser at 532 nm (green laser) was used as the light source with 60 mW laser power, a confocal hole of 300  $\mu\text{m}$ , slit size of 200  $\mu\text{m}$  and an exposure time of 120 s. This long acquisition time was preferred in order to improve signal-to-noise ratio, but no changes were observed in terms of either peak intensity or Raman shift with respect to test measurements conducted with shorter acquisition times. The Raman scattering was collected by a 100 $\times$  objective and  $\sim 5 \mu\text{m}^2$  spot size on the polished sample surface over a range from 100 to 1800  $\text{cm}^{-1}$  and from 2900 to 4200  $\text{cm}^{-1}$  with a resolution of 1.8  $\text{cm}^{-1}$ . Additional Raman spectroscopy measurements over the entire range 100–4000  $\text{cm}^{-1}$  using both green (532 nm) and red (632.8 nm) lasers were also performed in order to have a more detailed characterization of these peculiar quenched products. Spectral frequency was calibrated using the incident laser line and a Si single crystal. An average of 4 measurements were performed on each sample to verify its spectroscopic homogeneity. The obtained spectra were processed to account for the temperature and excitation line effects (Long, 1977) and PeakFit<sup>®</sup> software for smoothing, baseline removal and deconvolution.

### 2.5. Micro-infrared spectroscopy at both ambient and high pressure

The polished glasses embedded in epoxy resin were analyzed by Fourier transform micro-infrared spectroscopy in reflection mode ( $\mu$ -R-FTIR) at the Material Science branch of SISSI beamline of CNR-IOM at Elettra synchrotron in Trieste (Italy). The analyses were carried out in the region 600–6000  $\text{cm}^{-1}$  using a Vertex 70v interferometer coupled with a Hyperion2000 microscope (Bruker Optics), a nitrogen cooled HgCdTe (MCT) detector and a KBr beam splitter. Measurements were conducted using a conventional thermal source (Globar) with a resolution of 1  $\text{cm}^{-1}$  and an aperture of 80  $\mu\text{m}$  by collecting three acquisitions per area of 128 scans each. The reflectance was determined with respect to a flat gold mirror as it reflects near 100% of light in mid IR, the spectrum of which was collected under the same experimental conditions. Only CARB2\_4 sample was measured at the Department of Physics

of Sapienza University in Rome using the Vector 22 Bruker interferometer equipped with a broadband MCT detector and a Globar infrared source coupled to the Hyperion 1000 infrared microscope. For this sample, the reflectance spectra were collected between 600 and 8000  $\text{cm}^{-1}$  with a resolution of 2  $\text{cm}^{-1}$  with respect to a flat silver mirror ( $\sim$ 100% reflectivity in mid IR) using the same aperture from 50  $\mu\text{m} \times 50 \mu\text{m}$  to 200  $\mu\text{m} \times 200 \mu\text{m}$  in size. Three acquisitions of 128 scans each were performed on the selected area. In addition, transmission FTIR measurements ( $\mu$ -T-FTIR) were conducted on 200  $\mu\text{m}$ -thick double polished samples of CARB2\_2\_2016, CARB2\_2\_2018 and CARB2\_4 at the Department of Physics of Sapienza University in Rome using a Vertex 70v interferometer coupled to the Hyperion 1000 infrared microscope. Sample spectra as well as background spectra were collected between 400 and 6000  $\text{cm}^{-1}$  with a resolution of 2  $\text{cm}^{-1}$ . Five acquisitions of 128 scans each were collected.

The vibrational properties of CARB2\_4 glass were also investigated by  $\mu$ -R- and  $\mu$ -T-FTIR spectroscopy at HP and room T. A symmetric diamond anvil cell (DAC) was used for reflectance measurements coupled with the setup available at SISSI-Mat described above (Lupi et al., 2007; Piccirilli et al., 2012). A tiny chip of the glass ( $\sim$ 70  $\mu\text{m}$  of diameter) was loaded in the DAC equipped with low-fluorescence IIas-type diamonds with 500  $\mu\text{m}$  culet diameter, an Inconel gasket 100- $\mu\text{m}$  thick with a 200- $\mu\text{m}$  drilled hole, and KBr powder (transparent in the mid IR range) used as pressure medium. A ruby sphere was placed near the glass chip as for in situ pressure calibration (Mao et al., 1986). Mid-IR spectra were acquired in the range of 600–6000  $\text{cm}^{-1}$  with a resolution of 1  $\text{cm}^{-1}$  and an aperture of 60  $\mu\text{m}$ . Reflectance was calculated with respect to the intensity of Au corrected for the intensity reflected from the external face of the diamond.  $\mu$ -T-FTIR measurements were carried out at SISSI-MAT beamline with a Belleville spring DAC (LotoEng company) 20 mm height equipped with low-fluorescence IIas-type diamonds with 300  $\mu\text{m}$  culet diameter, an Inconel gasket 150- $\mu\text{m}$  thick with a 150- $\mu\text{m}$  drilled hole, using synchrotron light as source. Spectra were acquired in the range of 600–8000  $\text{cm}^{-1}$  with a resolution of 2  $\text{cm}^{-1}$  and an aperture of 30  $\mu\text{m}$  using CsI as quasi-hydrostatic P medium (transparent in the mid IR range). The characteristic spectra of either sample or diamond or salt were all collected with three acquisitions of 128 scans each. Transmittance was calculated by dividing the spectra of the sample in the DAC with CsI by those of the DAC filled with CsI only, which accounts for background contributions. The collected raw spectra were, then, elaborated using both OPUS<sup>TM</sup> (version 8.2) and ReFFIT softwares. Smoothing and deconvolution of the spectra were performed using PeakFit<sup>®</sup> software.

### 2.6. In situ Mössbauer analysis

The  $\text{Fe}^{3+}/\sum\text{Fe}$  ratio of CARB2\_4 glass (1.4 GPa/1815 °C) was determined at room T through in situ synchrotron Mössbauer source (SMS) spectroscopy at the ID18 Nuclear Resonance Beamline of the European Synchrotron Radiation Facility (ESRF; Grenoble, France). SMS technique exploits the (111) Bragg reflection of a  $^{57}\text{Fe}\text{BO}_3$  crystal, mounted on a Wissel velocity transducer and driven with a sinusoidal waveform, to obtain the Mössbauer characteristic energy of 14.4 keV (Potapkin et al., 2012). Kirkpatrick-Baez mirrors were used to focus the beam onto an area of about  $5 \times 15 \mu\text{m}^2$ . The velocity scale was set to  $\pm 6$  (mm/s) and calibrated using a 25- $\mu\text{m}$  thick  $\alpha$ -Fe foil. Data collection required 8 h, considering that the glass was not  $^{57}\text{Fe}$  enriched, with low FeO wt% (about 2.7 wt%). The spectrum was fitted by full transmission integral with a normalized pseudo-Voigt line-shape using the MossA software package (Prescher et al., 2012).

### 2.7. X-ray computed microtomography

X-ray computed microtomography (micro-CT) investigation was carried out on chips of samples CARB2\_1 (2.6 GPa/1865 °C) and CARB2\_4 (1.4 GPa/1815 °C) extracted from the C capsule using the Carl

Zeiss Xradia Versa-410 3D X-ray microscope available at the Istituto Nazionale di Geofisica e Vulcanologia-Sezione di Napoli “Osservatorio Vesuviano” (INGV-OV, Naples). The scan was performed over a 360° rotation using 1601 projections with a scan time of 18 s per projection, 40 kV voltage, and 3 W power with a spatial resolution of 1  $\mu\text{m}/\text{voxel}$ .

### 3. Results

#### 3.1. Liquid structure measurements at high pressure and temperature

We performed four successful measurements of the structure of carbonate-silicate liquids at P and T of 1.4 GPa/1815 °C (CARB2\_4), 2.6 GPa/1865 °C (CARB2\_1\_2016), 4.3 GPa/1990 °C (CARB2\_3\_2016) and 4.4 GPa/1950 °C (CARB2\_2\_2016) (Table 1). It can be noted that these P-T conditions are slightly different from those at which the viscosity was determined (e.g., CARB2\_2\_2016 and CARB2\_3\_2016; Stagno et al., 2020a). In fact, during liquid structure measurements, it is preferable to set T slightly higher than the liquidus T at which viscosity measurements are conducted to prevent partial crystallization of the liquid sample caused by any electrical power fluctuation during the 3-h long acquisition time. The increase of T also causes a change of P. We consider the  $P_{\text{fin}}$  and  $T_{\text{fin}}$ , which are estimated just before quench (Table 1) as representative of the target experimental conditions of the structure data. The typical uncertainties for PE experiments at HP-T are less than 0.4 GPa and 60 °C, as discussed in Kono et al. (2014a, 2014b).

The EDXD diffraction patterns of the molten samples collected at the 2 $\theta$  angle of 15.0161° just before data acquisition show their amorphous nature (some sharp peaks are from the graphite capsule, the surrounding ceramics and the WC anvils) at the investigated P-T conditions (Fig. S2a-d). Fig. 2 shows the structure factor,  $S(q)$ , determined for the carbonate-silicate liquid at high P-T with values reported in Table 1 for the first sharp diffraction peak (FSDP; Elliott, 1991) of  $S(q)$ . The differences among the patterns in Fig. 2 for each run can be explained as a result of P-T and composition. The position of the first sharp diffraction peak (FSDP; Elliott, 1991) in the structure factor  $S(q)$  was determined by Gaussian fitting and it varies from 2.13  $\text{\AA}^{-1}$  at 1.4 GPa/1815 °C to 2.25 at 4.3 GPa/1990 °C, showing a positive dependence with increasing the P and T. The FSDP, except for the spectrum at 4.3 GPa and 1990 °C, shows a shoulder at about 1.5–1.8  $\text{\AA}^{-1}$ . A second peak appears at  $\sim 3$   $\text{\AA}^{-1}$ , corresponding to the second sharp diffraction peak (SSDP; Elliott, 1995). The Fourier transformation of  $S(q)$  gives the reduced pair distribution function,  $G(r)$ , by which information about interatomic distances of T-O (T = Si, Al,  $\text{Fe}^{3+}$ ), T-T and M-O (M = Mg,  $\text{Fe}^{2+}$  and Ca) are usually determined. Here, conversely to studies conducted on either

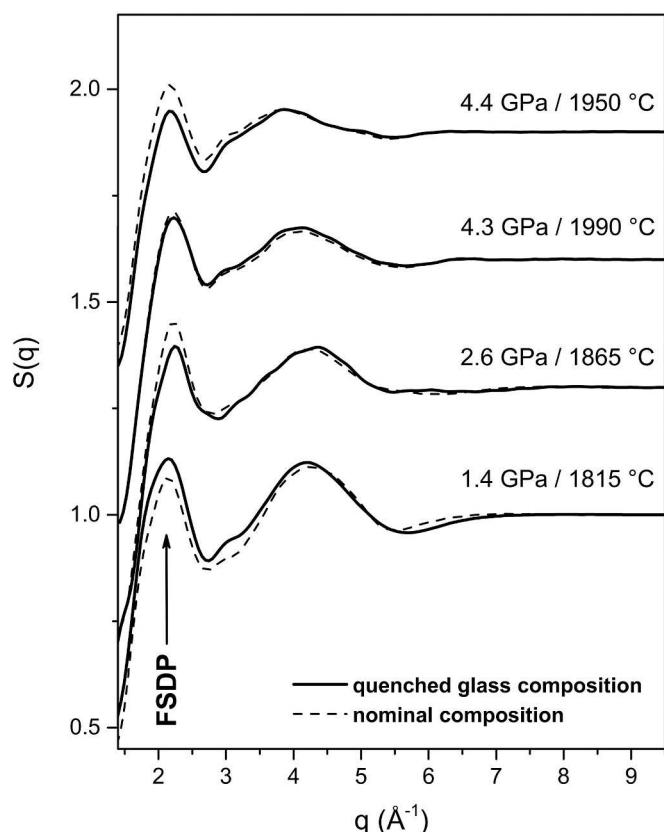


Fig. 2. Structure factor  $S(q)$  for CARB2 runs investigated by multi-angle energy dispersive X-ray diffraction at high pressure and temperature. Black lines refer to data processed using the composition of the quenched glasses (Table 2), while black dashed lines are representative of data processed using the nominal composition of the starting material. No significant differences are observed.

pure carbonate or silicate melts (e.g. Kono et al., 2014a; Hudspeth et al., 2018; Sakamaki et al., 2013; Wang et al., 2014), the chemical composition of the carbonate-silicate melt hampers the possibility to clearly distinguish such local structure of T (i.e., Si, Al and  $\text{Fe}^{3+}$ ) and M (i.e., Mg, Ca,  $\text{Fe}^{2+}$ ). This can be observed in Fig. S3 where the peak visible at 1.83–2.11  $\text{\AA}$  is indicative of both T-O and M-O contributions, the distinction of which is not straightforward given the complex composition of the melt investigated.

EDXD data were processed taking into account the chemical composition of the retrieved quenched glasses after the experiment (Table 2, see Figs. S4–S7 for additional information on data treatment). To assess any effect on structural parameters given by a variation in the melt chemical composition, we additionally processed data using the nominal composition of the starting material (Figs. S8–S11). The slight variations in  $S(q)$  between the two sets of data result in a negligible shift of the FSDP of  $\pm 0.02$   $\text{\AA}$ , as reported in brackets in Table 1 and shown in Fig. 2 (dashed lines).

#### 3.2. Chemical composition of the quenched glasses

The chemical composition of the five quenched glasses is reported in Table 2 with the P-T conditions at which the liquid structure measurements were carried out. Run CARB2\_2\_2016 is a glass recovered from a failed EDXD measurement. Photos taken under the optical microscope of the polished quenched glasses are reported in Fig. S1a–e. BSE images (Fig. S12a–e) show a vesicular glassy exposed sample surface. The chemical composition of the quenched products determined by EMP analysis varies from 18.75 to 30.91 wt% of  $\text{SiO}_2$ , 2.41 to 3.74 wt% of  $\text{Al}_2\text{O}_3$ , 28.06 to 41.47 wt% of  $\text{MgO}$ , 1.32 to 5.25 wt% of  $\text{FeO}$ , 9.82 to

Table 1

Experimental conditions for HP-HT liquid structure measurements and obtained structural data, determined processing EDXD data using the composition of recovered quenched glasses.

Run	${}^a\text{P}_{\text{initial}}$ (GPa)	${}^a\text{P}_{\text{final}}$ (GPa)	${}^a\text{T}_{\text{initial}}$ (°C)	${}^a\text{T}_{\text{final}}$ (°C)	FSDP ( $\text{\AA}^{-1}$ )	$\eta_{\text{calc}}$ (Pa·s)
CARB2_1_2016	2.8	2.6	1870	1865	2.25 (2.23)	0.0553
CARB2_2_2016	2.6	4.4	1990	1950	2.20 (2.18)	0.0414
CARB2_3_2016	4.5	4.3	2012	1990	2.25 (2.23)	0.0351
CARB2_4	1.3	1.4	1815	1815	2.13 (2.15)	0.0660

Notes: FSDP values in brackets refer to positions determined processing EDXD data using the nominal composition of the starting material, the uncertainty of which is discussed in detail by Kono et al. (2014a). Viscosities are calculated after data from Stagno et al., 2020a as discussed in Section 4.1 and Supplementary Materials. <sup>a</sup>Typical uncertainties on pressure and temperature in the Paris-Edinburgh cell experiments are less than 0.4 GPa and 60 °C (e.g., Kono et al., 2014a, 2014b; De Grouchy et al., 2017; Hudspeth et al., 2018).

**Table 2**

Chemical composition employed in the synthesis of CARB2 and results of EMPA analysis (INGV) on glasses quenched from HP-T experiments.

Oxides/samples	CARB2 nominal	CARB2_1_2016	CARB2_2_2016	CARB2_3_2016	CARB2_4	CARB2_2_2018
No. points analyzed	–	14	6	8	12	6
Quench P/T conditions	–	2.6 GPa 1865 °C	4.4 GPa 1950 °C	4.3 GPa 1990 °C	1.4 GPa 1815 °C	2.6 GPa 1735 °C
SiO <sub>2</sub>	18.28	23.09(47)	18.75(9)	22.02(50)	30.91(45)	24.95(73)
Al <sub>2</sub> O <sub>3</sub>	1.95	2.87(13)	2.41(7)	2.93(11)	3.74(13)	3.47(5)
FeO	6.72	1.57(06)	2.28(18)	1.32(10)	2.76(8)	5.25(6)
MgO	27.50	30.09(35)	28.06(27)	32.39(67)	41.47(21)	31.05(51)
CaO	20.43	15.79(13)	14.86(23)	16.71(28)	9.82(25)	15.55(30)
Na <sub>2</sub> O	1.37	0.64(11)	0.35(8)	0.77(7)	0.14(10)	0.79(36)
Cl <sup>–</sup>	1.21	6.35(33)	5.55(27)	4.80(13)	4.42(39)	4.98(70)
Vol	22.54*	19.54	27.72	18.99	6.68	13.09
Total	100.00	80.46(62)	72.28(73)	81.01(94)	93.32(54)	86.90(43)

Notes: values in brackets indicate 1-sigma uncertainty. \*CO<sub>2</sub> value of the starting material. Vol stays for the (C-O-H) volatile component dissolved in the quenched glass.

16.71 wt% of CaO, 0.14 to 0.79 wt% of Na<sub>2</sub>O, 4.42 to 6.35 wt% of ClO<sup>–</sup>. As a first approximation, the low totals (72.28–93.32 wt%) would indicate the presence of CO<sub>2</sub> that is lower than the CO<sub>2</sub>-rich starting material (22.54 wt% CO<sub>2</sub>) except for sample CARB2\_2\_2016 (27.72 wt% CO<sub>2</sub>). Such variation might result from the variable P-T conditions at which the liquids were exposed. On the other hand, Fig. S12a-e does show the occurrence of secondary graphite laths (other than the primary source of graphite, the capsule) along with (pseudo-)spherical vacuoles some of which are filled to suggest possible liquid unmixing of a calcitic component (see Supplementary Materials). These textural observations along with the tomographic images collected on samples CARB2\_1 and CARB2\_4 (see Movie in SM), showing the absence of exsolved gas bubbles, provide evidence of the CO<sub>2</sub> undersaturation conditions at the time of quench. Additional EMPA analyses conducted at INGV and BGI (Tables S1 and S2) confirm the chemical composition of the recovered quenched products with no evidence of the presence of undesired contaminants revealed by SEM.

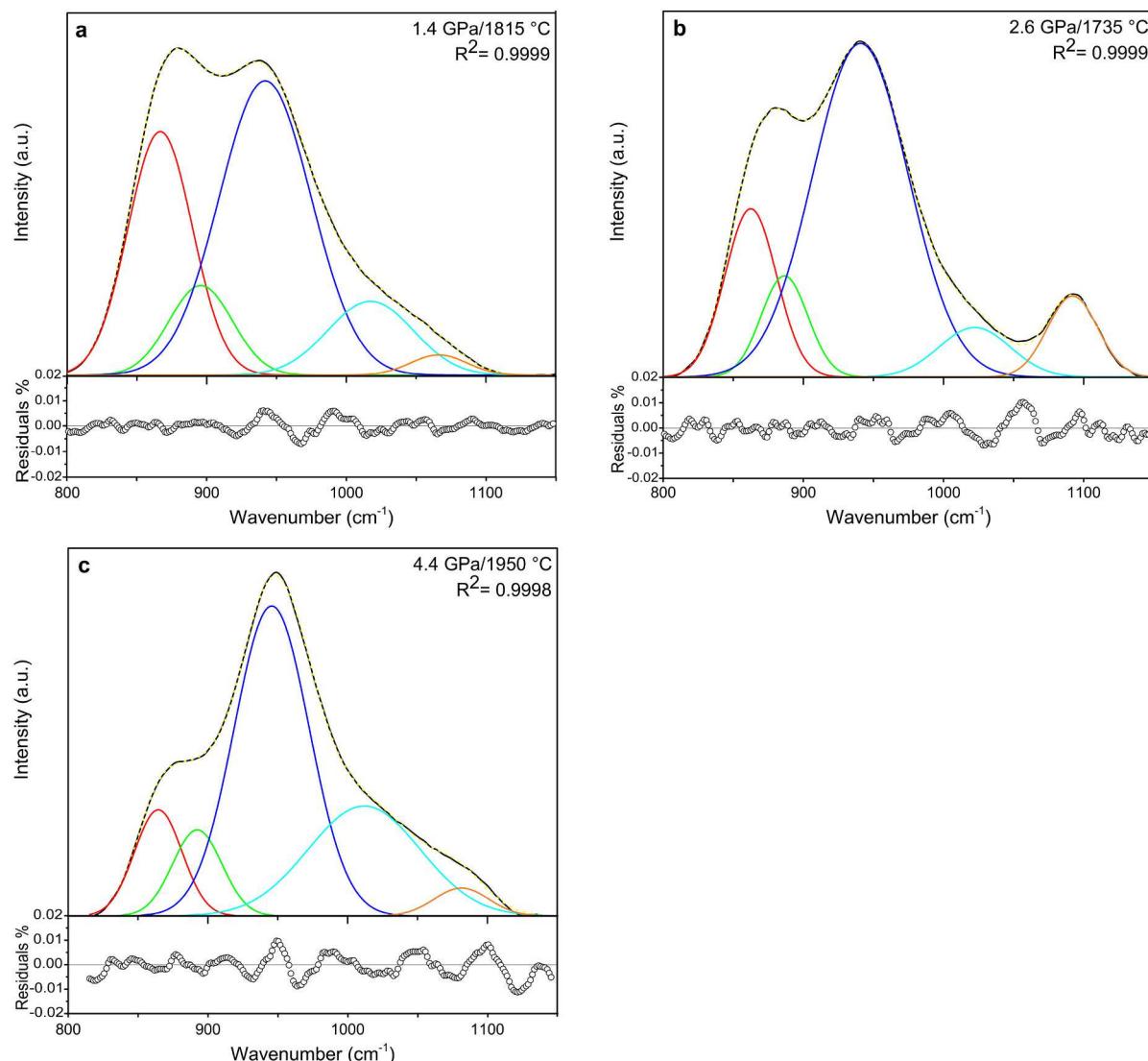
The lower FeO<sub>tot</sub> content observed in the quenched glasses (1.32–5.25 wt%) compared with that of nominal CARB2 (6.72 wt%) is due to the alloying with the falling Pt sphere (up to about 40 wt% of Fe) occurring during the duration of the liquid structure measurements (see Supplementary Materials), well after the descent of the Pt sphere for viscosity measurements. The content of CaO also decreased by ~5 wt% (~11 wt% in CARB2\_4; see Table 2) with respect to the nominal composition as a result of the segregation of some Ca-rich blobs of few microns in size formed during quench and surrounded by a Cl-rich thin halo (Fig. S12). In turn, a slight increase of SiO<sub>2</sub> is observed by ~4 and 5 wt% in run CARB2\_1\_2016 and CARB2\_3\_2016, respectively, and up to 12 wt% in CARB2\_4. Interestingly, the high MgO content (41.47 wt%) of CARB2\_4 glass is closely representative of carbonate-silicate magmas expected to form by melting of a carbonated harzburgite (Brey et al., 2008; Stagno and Frost, 2010; Girnis et al., 2011). As can be noticed in Table 2, the thin additional layers of CaCO<sub>3</sub> (run CARB2\_1\_2016) and San Carlos olivine+Li<sub>2</sub>CO<sub>3</sub> (run CARB2\_2\_2018) inserted in the experimental capsule for viscosity measurements did not result in any significant changes in chemical compositions of these quenched glasses when compared to the others with no additional layer. Despite the change in composition with respect to the nominal composition, our glasses fall in the field between carbonatites and natural kimberlites (Fig. 1), for which spectroscopic properties are unexplored.

### 3.3. Raman spectra of quenched glasses

Large pockets of homogenous whitish glass with dimensions of at least 10 μm x 10 μm are well distinguishable and exposed at the surface in all quenched glasses allowing the collection of confocal micro-Raman spectra on three samples CARB2\_2\_2016, CARB2\_4 and CARB2\_2\_2018. The other two samples (CARB2\_1 and CARB2\_3) could not be measured because of high fluorescence. The entire Raman spectra of the glasses

along with the spectrum of the employed epoxy resin in which experimental capsules were embedded are shown in Figs. S13-S15 of the Supplementary Materials. We observed no differences between spectra collected using green laser and red laser. These are characterized by bands distributed within two main regions of 700–1200 cm<sup>–1</sup> and 3100–3700 cm<sup>–1</sup>. In addition, Raman spectra of CARB2\_2\_2016 and CARB2\_4 glasses show small peaks arising in the region 2050–2150 cm<sup>–1</sup>. Spectra of carbonate-silicate glasses in the low frequency region are characterized by broad overlapped bands ranging from 850 to 1100 cm<sup>–1</sup>. These broad peaks arise from T-O stretching vibrations of Q<sub>n</sub>-species, where *n* refers to the number of bridging oxygens distributed among SiO<sub>4</sub> and AlO<sub>4</sub> units. The peak at 720 cm<sup>–1</sup>, plus a minor contribution at 790 cm<sup>–1</sup> in CARB2\_2\_2016 (Figs. S13-S15), has also been assigned to the Si-O stretching (translational) mode according to Matson et al. (1983). Given the composition of the starting material employed in this study, such convoluted bands are expected to contain also a contribution from the  $\nu_1$  symmetric stretching vibration of CO<sub>3</sub><sup>2–</sup> as it falls in this frequency range, too. Vibrations in the region 2050–2150 cm<sup>–1</sup> (Fig. S16) were attributed to carbon monoxide (Yoshioka et al., 2015, 2019; Armstrong et al., 2015; Grewal et al., 2020) both free and likely bonded to either Fe or Pt atoms to form carbonyls. The 3100–3700 cm<sup>–1</sup> interval is visible in runs CARB2\_2\_2016 and CARB2\_2\_2018 and assigned to the presence of OH bonds, which can be associated with the presence of either hydroxyl groups (OH) or molecular H<sub>2</sub>O. The amount of dissolved H<sub>2</sub>O was determined using the procedure proposed by Le Losq et al. (2012) that results in H<sub>2</sub>O content of 0.41 wt% and 0.16% wt% for CARB2\_2\_2016 and CARB2\_2\_2018, respectively. Similarly, the model proposed by Di Genova et al. (2017) results in 0.54 and 0.43 wt% H<sub>2</sub>O.

Fig. 3a-c shows the deconvoluted spectra at 800–1150 cm<sup>–1</sup> after correction for the temperature and excitation line effects using the equation proposed by Long (1977), smoothing and baseline subtraction using a cubic-spline line (anchor points at ~650, 785, 1150 cm<sup>–1</sup>). The best fit to our spectra was obtained using 5 peaks according to the vibrational modes as in Moussallam et al. (2016), representative for  $\nu_s$  Si-O in Q<sub>0</sub>, Q<sub>1</sub>, Q<sub>2</sub> silicate units + one peak for vibrations involving bridging oxygen atoms in any Q<sub>n</sub> ( $\nu_s$  Si-BO in Q<sub>n</sub>, hereinafter referred as Si-O<sup>0</sup>; MySEN et al., 1982, Le Losq et al., 2015) and one for  $\nu_1$  CO<sub>3</sub><sup>2–</sup>. The initial peak position and width were both left free to adjust until the optimal R<sup>2</sup> value was reached. All spectra show a peak at ~860 cm<sup>–1</sup> corresponding to Q<sub>0</sub> increasing in intensity from CARB2\_2\_2016 (2.41 wt% Al<sub>2</sub>O<sub>3</sub>) to CARB2\_2\_2018 (3.47 wt% Al<sub>2</sub>O<sub>3</sub>) and more intense in CARB2\_4 (3.74 wt% Al<sub>2</sub>O<sub>3</sub>). These are followed by a peak at 880 cm<sup>–1</sup> relative to Q<sub>1</sub> and at 950 cm<sup>–1</sup> (940 cm<sup>–1</sup> in CARB2\_2\_2018) representative of Q<sub>2</sub>. A peak at about 1020 cm<sup>–1</sup> corresponding to Si-O<sup>0</sup> appears as a shoulder to the dominant peak at 950 cm<sup>–1</sup> observed in all compositions. A deconvoluted peak at 1081 cm<sup>–1</sup> is observed in CARB2\_2\_2016 (27.72 wt% CO<sub>2</sub>) and a distinct peak in CARB2\_2\_2018 (13.09 wt% CO<sub>2</sub>) at 1092 cm<sup>–1</sup>, both attributable to  $\nu_1$  CO<sub>3</sub><sup>2–</sup> rather than any Q<sub>3</sub>



**Fig. 3.** Deconvoluted Raman spectra for runs CARB2\_4 (a), CARB2\_2 2018 (b) and CARB2\_2 2016 (c). Red, green, blue, light blue and orange fitting curves are for  $Q_0$ ,  $Q_1$ ,  $Q_2$ , Si-BO (see text) and  $\nu_1$  carbonate stretching, respectively. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

species of aluminosilicate units due to the weakly polymerized nature of our glasses. In CARB2\_4 (6.68 wt%  $\text{CO}_2$ ) this peak is less visible at 1066  $\text{cm}^{-1}$ . No evidence of molecular  $\text{CO}_2$  is observed in any of the three quenched glasses. Additional information of peak position, FWHM and area% are available in Supplementary Materials.

#### 3.4. $\mu$ -R- and $\mu$ -T-FTIR spectroscopy of the quenched glasses at ambient pressure

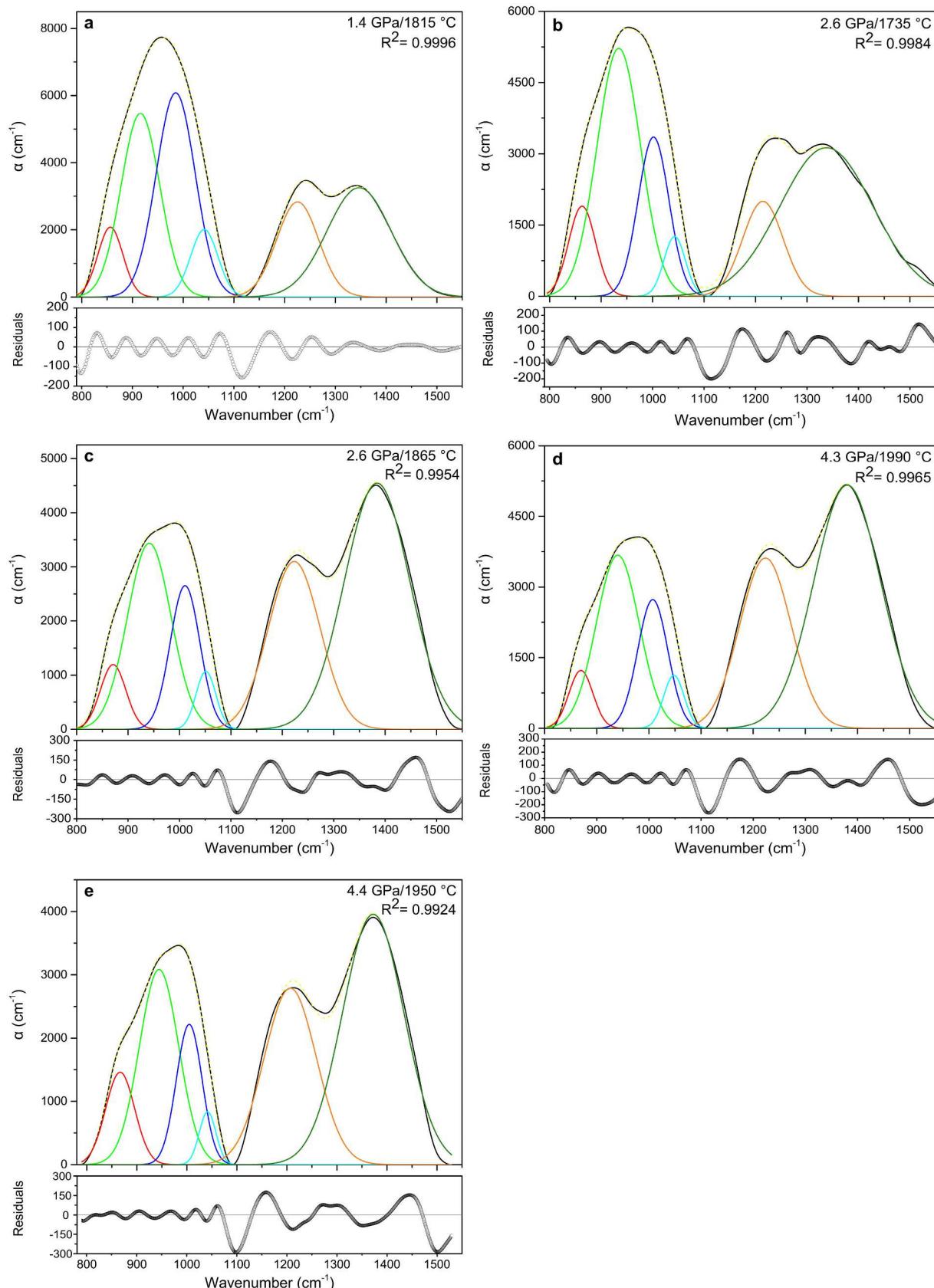
$\mu$ -R-FTIR spectra were collected from the quenched glasses listed in Table 2 embedded in epoxy resin. The full spectra are shown in Fig. S17 of the Supplementary Materials in the whole range of acquisition from 600 to 6000  $\text{cm}^{-1}$ , with the main features being visible within the 600–1500  $\text{cm}^{-1}$  and 3000–4000  $\text{cm}^{-1}$  regions. The assigned bands are after King et al. (2004) and Grzechnik et al. (1996). From the whole collected spectra, a mode at 660–695  $\text{cm}^{-1}$  arises and was associated to bending vibrations of C-O in  $\text{CO}_3^{2-}$ . Then, an intense broad band is observed at 940–980  $\text{cm}^{-1}$  that is related to the Si-O and Al-O asymmetric stretching vibrations in tetrahedral  $\text{TO}_4$ - $\text{T}_2\text{O}_7$ - $\text{TO}_3$  orthosilicate-pyrosilicate-metasilicate units. The doublet at  $\sim 1210$ –1380  $\text{cm}^{-1}$  is representative of the  $\nu_3$  asymmetric stretching vibrations of carbonate

groups, and results in an increase of the splitting ( $\Delta\nu_3$ ) from 121  $\text{cm}^{-1}$  (CARB2\_4) to 165  $\text{cm}^{-1}$  (CARB2\_2\_2016), which appears to be correlated to the dissolved volatile content. In addition, the carbonate doublet appears asymmetric, with the peak at lower wavenumber being more intense from 1.4 GPa/1815 °C (run 4) to 4.4 GPa/1950 °C (run 2\_2016) and 4.3 GPa/1990 °C (run 3\_2016). No evidence of molecular  $\text{CO}_2$  is observed in any of the FTIR spectra obtained. The presence of the band near 3550  $\text{cm}^{-1}$  is representative of  $\text{H}_2\text{O}_{\text{tot}}$  that likely entered the sample during the experiment of melt structure or during the loading of the capsule.

From each spectrum the absorption coefficient ( $\alpha$ ) of the carbonate-silicate glasses as a function of frequency was determined using the following equation,

$$\alpha(\nu) = 4\pi\nu k \quad (1)$$

where  $\nu$  is the frequency in  $\text{cm}^{-1}$  and  $k$  the imaginary part of the refractive index calculated along with the real part,  $n$ , (McMillan, 1985) using the Kramers-Kronig (KK) transformation of reflectance data as part of the OPUS™ software package. Fig. 4a-e displays the absorption coefficient  $\alpha$  in the region 800–1550  $\text{cm}^{-1}$  derived from KK transformation of reflectance spectra, after baseline correction, smoothing and



**Fig. 4.** Deconvoluted FTIR spectra with absorption coefficient  $\alpha$  calculated from reflectance data using the Kramers-Kronig analysis. Red, green, blue lines correspond to  $Q_0$ ,  $Q_1$ ,  $Q_2$  species, respectively. The other lines correspond to carbonate vibrations  $\nu_1$  (light blue) and  $\nu_3$  doublet (orange and dark green lines). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

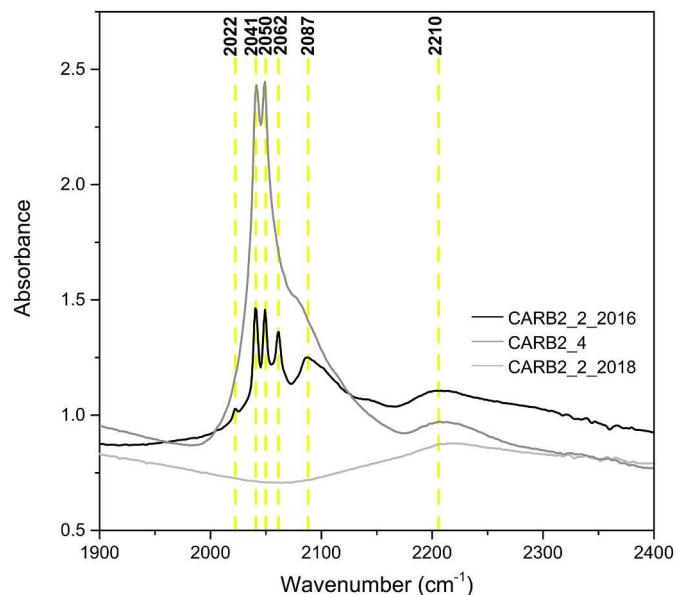
deconvolution. We employed 6 peaks positioned at  $\sim 860\text{ cm}^{-1}$ ,  $910\text{ cm}^{-1}$ ,  $960\text{ cm}^{-1}$ ,  $1040\text{ cm}^{-1}$ ,  $1220\text{ cm}^{-1}$ ,  $1360\text{ cm}^{-1}$  taking as reference literature data (Pirou and Arashi, 1980; McMillan and Wolf, 1995; Genge et al., 1995; King et al., 2004; Wilding et al., 2019) to account both for the silicate and carbonate contributions. Peak parameters were not constrained during iteration to achieve the best  $R^2$  value. Results from the deconvolution procedure carried out in the silicate region for CARB2 glasses show the four peaks centered between  $856$  and  $871\text{ cm}^{-1}$  (I),  $916$ – $944\text{ cm}^{-1}$  (II),  $985$ – $1011\text{ cm}^{-1}$  (III),  $1041$ – $1051\text{ cm}^{-1}$  (IV),  $1207$ – $1226\text{ cm}^{-1}$  (V),  $1337$ – $1384\text{ cm}^{-1}$  (VI). We assign bands I-III to the asymmetric stretching vibrations of  $Q_n$  tetrahedral units in  $\text{TO}_4$  units ( $Q_0$  species),  $\text{T}_2\text{O}_7$  ( $Q_1$  species) and  $\text{TO}_3$  units ( $Q_2$  species), respectively. The band IV was assigned to  $\nu_1$  symmetric stretching of  $\text{CO}_3^{2-}$ . Bands V and VI were assigned to the  $\nu_3$  asymmetric stretching vibrations of the carbonate doublet. A shift to higher wavenumbers is observed for the silicate bands I, II and III as a function of decreasing  $\text{SiO}_2 + \text{Al}_2\text{O}_3$  content and increasing volatile content (Table 2), whereas the opposite trend is observed for band V.

The  $\mu$ -T-FTIR spectra were acquired on double-polished,  $200\text{ }\mu\text{m}$ -thick samples CARB2\_2\_2016, CARB2\_4, CARB2\_2\_2018. Transmittance spectra are shown in Fig. 5 to underline the saturation (i.e. transmittance value equal to zero) observed from  $400$  to  $1600\text{ cm}^{-1}$ , therefore testifying the carbonated nature of the quenched glasses.

In order to visualize the carbonate bands, the quenched glasses were required to be double polished to quite low thickness. However, the combination of  $\mu$ -T-FTIR to  $\mu$ -R-FTIR allowed the successful resolution of this spectral region as illustrated previously.

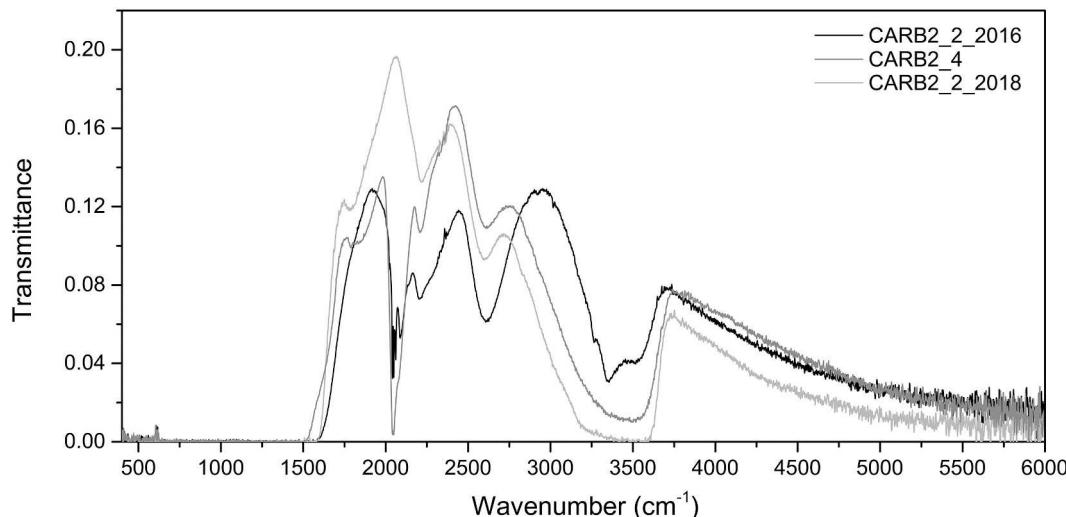
Consistent with that which is observed by Raman spectroscopy in Fig. S16, the  $2000$ – $2100\text{ cm}^{-1}$  region shows vibrations at  $2022$ ,  $2041$ ,  $2050$ ,  $2062$  and  $2087\text{ cm}^{-1}$  (Fig. 6) in CARB2\_2\_2016 and CARB2\_4 (some occurring as a shoulder in the latter). Those peaks are assigned to C-O stretching vibrations associated with carbonyl groups bonded to a transitional metal (Xu, 2002 and references therein). Further, a peak at  $2210\text{ cm}^{-1}$  attributed to free molecular carbon monoxide (Eggler et al., 1979; Brooker et al., 1998, 1999; Stanley et al., 2014; Armstrong et al., 2015; Yoshioka et al., 2015, 2019) is observed in all three samples investigated, which is shifted with respect to the fundamental vibration of gaseous CO at  $2143\text{ cm}^{-1}$  as CO is dissolved in the glass matrix, in agreement with results reported in the aforementioned studies.

All three of the investigated samples show absorptions in the  $3000$ – $3700\text{ cm}^{-1}$  region (Fig. 5), which is relative to O-H stretching. CARB2\_2\_2016 shows more pronounced peaks at  $3265$  and  $3355\text{ cm}^{-1}$ . An estimation of dissolved  $\text{H}_2\text{O}_{\text{tot}}$  from transmission measurements of



**Fig. 6.** Absorbance spectra of CARB2 quenched glasses showing peaks attributed to metal carbonyl complexes in CARB2\_2\_2016 and CARB2\_4 at  $2022$ ,  $2041$ ,  $2050$ ,  $2062$  and  $2087\text{ cm}^{-1}$ , along with free carbon monoxide ( $2210\text{ cm}^{-1}$ ) which is instead present in all of the three quenched products.

the quenched glasses is proposed by applying the Lambert-Beer law using a range of values of extinction coefficients available in literature for mafic glasses, which vary from  $59$  to  $68\text{ L/mol}\cdot\text{cm}$  (Shishkina et al., 2014 and references therein) and a density interval of  $2300\text{ g/L}$  (Ghosh and Karki, 2017) and  $2600\text{ g/L}$ . Our estimation of  $\text{H}_2\text{O}_{\text{tot}}$  yielded values comprised between  $0.2$  and  $0.3\text{ wt\%}$  for CARB2\_2\_2016 and  $0.5$ – $0.6\text{ wt\%}$  for CARB2\_4. As it can be observed in Fig. 5, the transmission spectra collected for CARB2\_2\_2018 show saturation also in the O-H region. Therefore, based on our data we can estimate a  $\text{H}_2\text{O}_{\text{tot}}$  content which is expected to be slightly higher than  $0.8$ – $1\text{ wt\%}$  for CARB2\_2\_2018 glass. It is worth noting that it is extremely difficult to keep these systems dry, as inferred by Weidendorfer et al. (2020). However, it is likely that such low amount of water only has a negligible effect on both viscosity and liquid structure considering their depolymerized nature.



**Fig. 5.** Mid-infrared transmission spectra of recovered CARB2\_2\_2016, CARB2\_4 and CARB2\_2\_2018 glasses collected at ambient pressure and room temperature conditions.

### 3.5. $\mu$ -R-FTIR and $\mu$ -T-FTIR spectroscopy of a carbonate-silicate glass at high pressure

The effect of P was investigated on chips of transparent glass extracted from sample CARB2\_4. Spectra were collected in reflection mode during compression up to 5.6 GPa and decompression and analyzed for changes both in peak intensity (reflectance) and spectral shift. A procedure similar to that adopted for ambient P spectra was employed in order to better understand each specific contribution arising both from carbonate and silicate vibrations. Refractive indexes were obtained from the reflectance spectra taking into account the effect of diamond. Then, absorption coefficients were calculated using Eq. (1), and the same procedure as for quenched glasses in air was adopted. Fig. 7 shows the absorption coefficient,  $\alpha$ , calculated from reflectance spectra (with Au used as reference) after removal of sample-diamond interface contribution. The ambient P spectrum shows a broad band in the spectral regions 800–1160  $\text{cm}^{-1}$  associated to silicate modes, and an absorption doublet around 1180–1270  $\text{cm}^{-1}$ , we associated to carbonates. Lower pressure spectra show two evident peaks at 970  $\text{cm}^{-1}$  and 1030  $\text{cm}^{-1}$  composing the band. When pressure is increased, the peak at 1030  $\text{cm}^{-1}$  becomes more intense, as can be observed from spectra collected at 2.4, 4.1 and 5.6 GPa, possibly causing the gradual blueshift of the whole band to 1090  $\text{cm}^{-1}$ . This shift is reversible upon decompression as can be seen in Fig. 7; while a negligible variation is observed for the carbonate doublets. To achieve a quantitative structural evaluation, we used a deconvolution procedure based on peaks assignment described above. Importantly, an additional peak placed at 1080–1100  $\text{cm}^{-1}$  was necessary to fit all spectra (Fig. S18), in contrast with those at ambient P discussed above. This peak was attributed to the formation of  $\text{Q}_3$  species for increasing pressure at room T. The position of the first peak shows no significant spectral shift with increasing P, the second peak was observed to shift from 911  $\text{cm}^{-1}$  to 916  $\text{cm}^{-1}$  with increasing P and from 915  $\text{cm}^{-1}$  to 906  $\text{cm}^{-1}$  upon decompression. The carbonate

doublet shifts to lower wavenumbers with respect to ambient-P data upon compression with the peaks varying from 1188 to 1180  $\text{cm}^{-1}$  and from 1270 to 1255  $\text{cm}^{-1}$  up to 4.1 GPa. At 5.6 GPa the doublet shifts to higher frequencies as 1196 and 1260  $\text{cm}^{-1}$ . In addition, the splitting between the two peaks decreases from 82 to 64  $\text{cm}^{-1}$  with increasing P.

Spectra relative to in situ HP  $\mu$ -T-FTIR investigation of CARB2\_4 glass chip from 0.8 up to 11.1 GPa are shown in Fig. S19, as well as that of the retrieved decompressed sample. In these spectra peaks at  $\sim$ 720, 1080, 1460, 2900, 4250, 4320, 5680 and 5780  $\text{cm}^{-1}$  can be distinguished. These peaks were assigned to  $\text{CO}_3^{2-}$  vibrations (e.g. Genge et al., 1995), further confirming the high carbonate content of the quenched glasses. Specifically,  $\text{CO}_3^{2-}$   $\nu_4$  bending is observed at 720  $\text{cm}^{-1}$ , 1080  $\text{cm}^{-1}$  band was assigned to  $\nu_1$  stretching and 1460  $\text{cm}^{-1}$  to  $\nu_3$  stretching, with overtones 2 $\nu_3$ , 3 $\nu_3$  and 4 $\nu_3$  at 2900  $\text{cm}^{-1}$ , 4250 and 4320  $\text{cm}^{-1}$ , 5680 and 5780  $\text{cm}^{-1}$ , respectively. These peaks show saturation in transmission up to 2900  $\text{cm}^{-1}$  (Fig. S19a), hindering the possibility to precisely determine band positions and how these vary during compression at room T, as well as to resolve  $\nu_3$  stretching doublet (i.e. it appears as one band only), which is instead visible for  $\text{CO}_3^{2-}$  3 $\nu_3$  and 4 $\nu_3$  overtones (Fig. S19c). The region 1800–2700  $\text{cm}^{-1}$  is hidden by diamond anvils, therefore covering any free CO and/or carbonyl contributions which were instead observed at aP (Figs. 6 and 7). As P is increased, a blueshift occurs for all bands (Fig. S19 b, c), accompanied by broadening.

### 3.6. Iron oxidation state of carbonate-silicate glass

The  $\text{Fe}^{3+}/\sum\text{Fe}$  of CARB2\_4 was determined by in situ SMS spectroscopy. The choice of this quenched glass was motivated by the marked spectral features of carbonyls (i.e., higher peak intensity) and, therefore, the possibility to investigate a possible redox mechanism involving the fluid species and the liquid at high P and T. The collected spectrum was fitted with two symmetric doublets, one for  $\text{Fe}^{2+}$  and one for  $\text{Fe}^{3+}$  (Fig. S20). The hyperfine parameters used for the fitting, quadrupole splitting (QS), central shift (CS) and full width at half maximum (FWHM) are reported in Table S3 and are compatible with literature data for basaltic glasses (e.g., Partzsch et al., 2004; Blundy et al., 2020). The contribution of Fe from the confocal Be-lenses, a component of the beamline set-up, was constrained by fitting the spectrum with an extra singlet (CS = 0.2991 mm/s; FWHM = 0.5694), based on preliminary background measurements. The  $\text{Fe}^{3+}/\sum\text{Fe}$  ratio resulted in  $0.14 \pm 0.03$ , which is lower than what was expected in the case of  $\text{CO}_2$ -bearing oxidized liquids (and glasses).

## 4. Discussion

### 4.1. The structure of $\text{CO}_2$ -bearing melts with implications for magma viscosity

Table 1 shows the position of the FSDP from structure factors determined from EDXD measurements at different P/T conditions. The structural investigation conducted by EDXD at high P and T showed that carbonate-silicate liquids are characterized by a shift in the FSDP position in reciprocal space towards higher values as a result of the combined effect of increasing both P and T. As a consequence, the intermediate range order (IRO =  $2\pi/q_{\text{FSDP}}$ ) of carbonate-silicate liquids undergoes a shrinkage (from 2.95  $\text{\AA}$  to 2.79  $\text{\AA}$ ), in agreement with observations for molten calcite (Hudspeth et al., 2018), melilititic liquids (Stagno et al., 2020b) and silicate liquids with different degrees of polymerization (e.g. diopsidic and jadeitic liquids discussed by Wang et al., 2014). In contrast, the position of the FSDP extrapolated from the structure factor reported for MORB liquids by Sakamaki et al. (2013) gives a negative correlation as a function of P (and T), peculiar among silicate compositions. This behavior moreover is opposite to that observed upon cold compression for glasses with similar composition (Ohashi et al., 2018).

Our results allow exploration of the dependence of the viscosity of

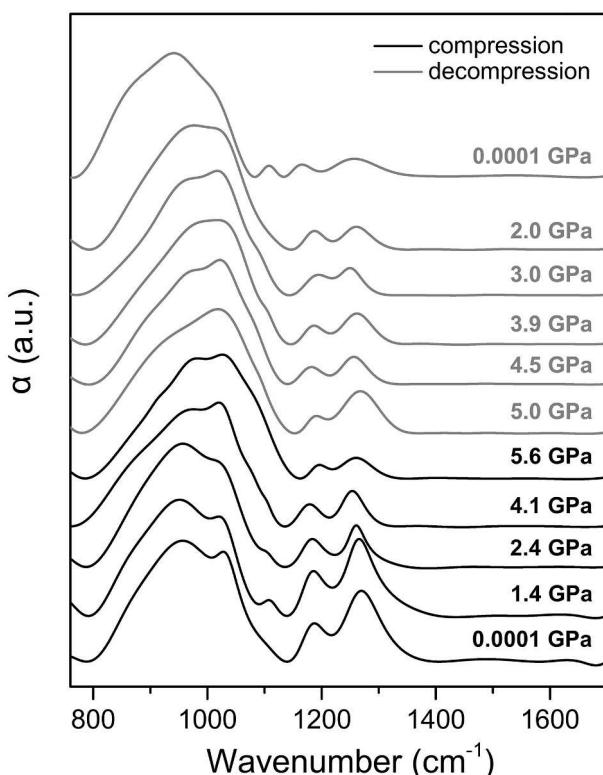


Fig. 7. The absorption coefficient  $\alpha$  calculated from reflectance FTIR spectra collected upon compression and decompression on CARB2\_4 glass at room T and high P.

carbonate-silicate liquids at high P-T on their structure and compare this with the data available for calcite (Kono et al., 2014b; Hudspeth et al., 2018), MORB (Sakamaki et al., 2013) and melilititic liquids (Stagno et al., 2020b). Viscosity data for carbonate-silicate liquids (Stagno et al., 2020a) were fitted using a Non-Arrhenian Vogel-Fulcher-Tammann (VFT) equation modified to account for the effect of pressure as in Ritter et al. (2021), details can be found in Supplementary Materials.

Fig. 8 shows the viscosity calculated at the P and T at which the FSDP is measured for each composition. Calcitic liquids are characterized by a constant viscosity over the range of FSDP suggesting that variations in the medium range order such as shrinkage and packing of the carbonate molecules does not necessarily imply changes in the transport properties of these melts. In contrast, the incorporation of Si atoms represented by carbonate-silicate liquids contributes to the increase of polymerization accompanied by an increase in viscosity. Liquids with 22–30 wt% SiO<sub>2</sub> are characterized by viscosities up to 1 order of magnitude higher with respect to calcitic (but also dolomitic) liquids. The increase of SiO<sub>2</sub> (and Al<sub>2</sub>O<sub>3</sub>) in the melilititic liquid (Stagno et al., 2020b) causes the viscosity to further increase. Because T acts to stretch out the Si(Al)-coordinated polyhedra, the polymerized network undergoes relaxation associated with a viscosity decrease. In the case of carbonate-silicate and foiditic liquids, the decrease in IRO also causes the viscosity to decrease, a behavior that appears opposite to that observed in the case of more polymerized MORBs (Sakamaki et al., 2013).

These experimental observations raise important considerations on whether the viscosity of CO<sub>2</sub>-bearing magmas at upper mantle P-T conditions should be treated by taking into account the amount of dissolved carbonate versus SiO<sub>2</sub> (Morizet et al., 2017b) or by implementing available current models that take into account also the melt structure as function of P-T and chemical composition.

#### 4.2. Vibrational properties of carbonate-silicate glasses from high P-T experiments with implications for carbon speciation in the upper mantle

The sparse reflectance FTIR data for CO<sub>2</sub>-bearing silicate glasses differ strongly from the glasses investigated in this study. Basanitic and leucitic glasses (Grzechnik et al., 1996), rhyolitic glasses (Moore et al., 2000), and melilititic glasses (Stagno et al., 2020b) are all characterized by two main regions of Si(Al)-O stretching vibrations (900–1100 cm<sup>-1</sup>) and molecular/ionic C (CO<sub>2</sub>, 2350 cm<sup>-1</sup>, CO<sub>3</sub><sup>2-</sup>, ~1300–1500 cm<sup>-1</sup>). Features in the silicate region shift to lower wavenumber by increasing the CO<sub>3</sub><sup>2-</sup> content into the melt as a result of depolymerization of the

silicate network. Our spectra are characterized by silicate peaks at even lower frequencies than those reported by the aforementioned literature, which denotes the extreme depolymerized nature of these carbonate-silicate glasses. As far as the fingerprint relative to the carbonate anion is concerned, we observe a strong influence of the chemical composition in the vibrational properties of ν<sub>3</sub> asymmetric stretch of CO<sub>3</sub><sup>2-</sup> where α (here indicated in terms of midpoint) decreases in wavenumber from ~1480 cm<sup>-1</sup> in leucite-basanite glasses (Grzechnik et al., 1996) to 1315 cm<sup>-1</sup> and 1306 cm<sup>-1</sup> in melilititic glasses (Stagno et al., 2020b) to about 1300–1285 cm<sup>-1</sup> as observed for carbonate-silicate glasses in this study. Such a marked shift was neither reported in the reflectance spectra of SiO<sub>2</sub>-free carbonate-rich glasses (Genge et al., 1995) nor in previous studies carried out in transmission mode (Moussallam et al., 2016). Importantly, the dependence of the shift to lower wavenumbers of the carbonate doublet from the chemical composition is also accompanied by an increase in doublet splitting, which varies from ~90 cm<sup>-1</sup> (leucite basanite glasses, Grzechnik et al., 1996) to 130 cm<sup>-1</sup> and 140 cm<sup>-1</sup> for melilititic glasses (Stagno et al., 2020b) up to 165 cm<sup>-1</sup> in the compositions investigated in this study (Fig. 4). The increase in doublet splitting is related to an increase in the degree of distortion of CO<sub>3</sub><sup>2-</sup> from the ideal trigonal planar symmetry (e.g. Blank and Brooker, 1994; Brooker et al., 2001b). In contrast to the Raman spectra, the reflectance spectra and the derived absorption coefficients clearly show the increase of the relative intensity between carbonate and silicate groups as the CO<sub>2</sub> content increases. In particular, the spectra taken on glasses CARB2\_1 and CARB2\_2\_2016 present the carbonate doublets (ν<sub>3</sub>) more pronounced with respect to the silicate group, opposite to what has been observed by Raman spectroscopy where the contribution of ν<sub>1</sub> stretching of CO<sub>3</sub><sup>2-</sup> appears to be weak with respect to the silicate vibrations. Noteworthy, this ν<sub>1</sub> stretching vibration was also identified in our reflectance data (1040–1051 cm<sup>-1</sup>), along with the bending vibration of C-O in CO<sub>3</sub><sup>2-</sup> (660–695 cm<sup>-1</sup>), which becomes more and more pronounced with increasing volatile content as determined from EMPA totals (Fig. S17).

The coexistence of CO<sub>3</sub><sup>2-</sup> with CO (either interpreted as free CO or CO linked to a transition metal, i.e. Fe) was previously observed in studies of the CO solubility in silicate melts and the speciation of C species as a function of oxygen fugacity (Eggler et al., 1979; Brooker et al., 1998, 1999; Stanley et al., 2014; Armstrong et al., 2015; Yoshioka et al., 2019). The FTIR peak located between ~2100–2200 cm<sup>-1</sup> has been previously assigned to free CO present in glasses (e.g. Eggler et al., 1979; Brooker et al., 1998, 1999), whereas Wetzel et al. (2013) and Stanley et al. (2014) proposed the formation of Fe-carbonyl complexes. Later works (Yoshioka et al., 2015, 2019) stressed that such peaks occurring around 2110 cm<sup>-1</sup> in Raman spectra and 2210 cm<sup>-1</sup> in FTIR spectra still appeared in CO-bearing Fe-free glasses, questioning their attribution to Fe-carbonyls. Metal carbonyl absorption bands usually occur between 1800 and 2100 cm<sup>-1</sup>, with the position depending on the bonding mode (e.g., terminal, metal-CO, or bridging, metal-CO-metal), metal involved and electron density (Zhou et al., 2001; Socrates, 2004). Considerable literature on CO adsorption on metals, metal oxides and halides as well as zeolites (as reviewed by Xu, 2002) showed that a key feature of carbonyls is a band centered between 2100 and 2000 cm<sup>-1</sup> made of a series of infrared absorptions (~4) occurring with an interval of 10–35 cm<sup>-1</sup> one from another. Studies from various fields are helpful on pointing out that the different compounds present in the system under investigation play a role in determining the position of peaks with respect to those of metal carbonyls alone. For instance, Hugues et al. (1982) investigated the effect of Fe<sub>3</sub>(CO)<sub>12</sub> and Fe(CO)<sub>5</sub> adsorbed on SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub> and MgO, revealing bands at 2013, 2033, 2053, 2110 cm<sup>-1</sup> in the case of SiO<sub>2</sub>, 2008, 2020, 2082 cm<sup>-1</sup> on hydrated Al<sub>2</sub>O<sub>3</sub>, 2010 and 2078 cm<sup>-1</sup> on MgO. Trusheim and Jackson (1983) investigated SiO<sub>2</sub>-adsorbed Fe(CO)<sub>5</sub> finding peaks at 2002 and 2024 cm<sup>-1</sup> and Fe(CO)<sub>5</sub> coexisting with Fe<sub>3</sub>(CO)<sub>12</sub> under certain conditions, identified from a peak at 2052 cm<sup>-1</sup>. Brandt et al. (1993) studied CO adsorbed on Pt/SiO<sub>2</sub> determining absorptions at 2065, 2080, 2087 cm<sup>-1</sup>, slightly

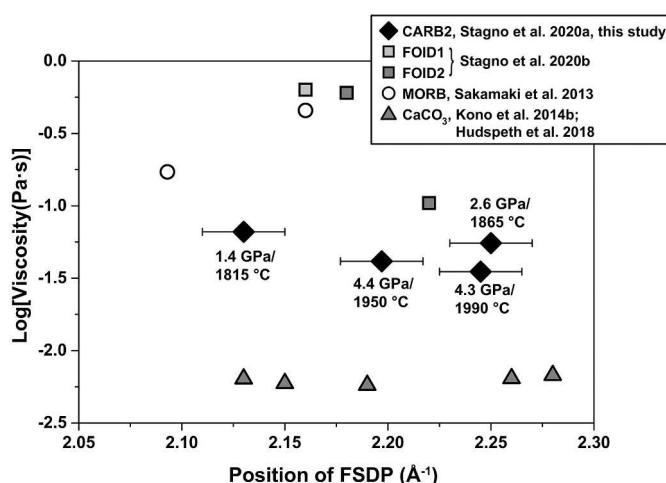


Fig. 8. Dependence of viscosity on the position of the first sharp diffraction peak (FSDP) in the structure factor for carbonate-silicate liquids (this study, Stagno et al., 2020a), molten calcite (Hudspeth et al., 2018; Kono et al., 2014b), melilititic liquids (Stagno et al., 2020b) and MOR basaltic liquids (Sakamaki et al., 2013).

shifting depending on particle size, and Stakheev et al. (1995) determined that CO adsorbed on Pt/KL zeolite resulted in the formation of Pt carbonyl peaks at 2008, 2031 and 2068  $\text{cm}^{-1}$ . Therefore, the presence of absorptions at 2022, 2041, 2050, 2062, 2087  $\text{cm}^{-1}$  in CARB2.2\_2016 and CARB2.4 (some occurring as shoulders rather than distinct peaks in the latter) led us to assign those bands to metal carbonyl complexes were Fe, Pt or a PtFe alloy may act as the CO ligand, coexisting with free CO as testified by the peak at 2210  $\text{cm}^{-1}$  (Fig. 6). Likewise, carbonyls (+free CO) might be recognized also in the small peaks of Raman spectra at 2060, 2077 and 2145  $\text{cm}^{-1}$ , with free CO at 2100  $\text{cm}^{-1}$  (Fig. S16).

The investigation of Ca-rich carbonate-silicate glasses synthesized under fully oxidized conditions by Raman spectroscopy (Moussallam et al., 2016) showed that the region of (Si-Al)-O stretching vibration is mostly dominated in order by  $Q_0 > Q_1 > Q_2 > \text{Si-O}^0$  along with an intense and sharp peak  $\nu_1\text{CO}_3^{2-}$  at  $\sim 1080 \text{ cm}^{-1}$  attributed to C-O stretching. This general spectral profile for silicate units is observed for glasses with 24 wt%  $\text{SiO}_2$  and variable  $\text{CO}_2$  contents, as well as for glasses with 31–32 wt %  $\text{SiO}_2$  and no  $\text{CO}_2$ . The increase of  $\text{CO}_2$  content is marked by the increase in intensity of  $Q_1$  at 10 wt%  $\text{CO}_2$ , and  $\text{Si-O}^0$  at 18 wt%  $\text{CO}_2$ . In contrast, an estimate of the abundance of  $Q_n$  species obtained through IR peaks relative areas indicates that  $Q_2$  and  $Q_0$  are the most abundant species in CARB2 quenched glasses, and that the vibrations of  $\text{CO}_3^{2-}$  for  $\nu_1$  contribution is much less active with respect to the data from Moussallam et al. (2016). In particular, glasses with  $\sim 19$  and 23 wt%  $\text{SiO}_2$  and estimated volatile content of 28 and 19 wt%, respectively, have a higher abundance of  $Q_2$  than  $Q_0$  and  $Q_1$  with  $\text{Si-O}^0$  being more pronounced at higher volatiles content. The glass with  $\sim 31$  wt%  $\text{SiO}_2$  and  $\sim 7$  wt% volatiles is characterized by intensities that vary with  $Q_2 > Q_0 > Q_1 > \text{Si-O}^0$  with a tiny contribution of  $\nu_1\text{CO}_3^{2-}$ . On the other hand, we cannot exclude that the increasing intensity of  $\text{Si-O}^0$  in the glass with high  $\text{CO}_2$  content (CARB2\_2016) results from the shift of the  $\nu_1\text{CO}_3^{2-}$  band to lower wavenumbers pointing out, therefore, differences of C in bonding within the structure. A mechanism of  $\text{CO}_3^{2-}$  speciation by the formation of metal-carbonate frameworks deriving from carbonate ions strongly interacting with metal cations was proposed for the case of carbonate glasses by Genge et al. (1995) and reported to cause a shift in wavenumber. This could also occur in our glasses. Alternatively, the low intensity of  $\nu_1\text{CO}_3^{2-}$  despite the high  $\text{CO}_2$  content, which is testified by the prominent features of  $\text{CO}_3^{2-}$  bending and stretching vibrations in  $\mu$ -R-FTIR spectra, saturation of  $\mu$ -T-FTIR spectra and low EMPA totals, can be explained as due to the peculiar glass composition and experimental conditions which may affect carbonate polarizability, and for which we failed to employ the quantification model proposed by Morizet et al. (2013).

High P  $\mu$ -R-FTIR investigation showed that as the quenched glasses undergo compression, the asymmetry of  $\text{CO}_3^{2-}$  is observed to decrease as testified by the decrease in splitting observed in the carbonate doublet from  $\mu$ -R-FTIR and  $\mu$ -T-FTIR (visible in  $3\nu_3$  and  $4\nu_4$  overtones, Fig. S19c), implying a rearrangement of the structure where the carbonate environment gradually becomes more ordered and the silicate network more packed.

Finally, the spectroscopic evidence of molecular carbonate coexisting with carbon monoxide, water and carbonyl points out the possibility that these volatile species can be all dissolved in  $\text{CO}_2$ -rich magmas and be exsolved upon decompression. Such evidence would require the investigation of similar glasses quenched at lower T than those of this study or the possibility to investigate spectroscopically similar liquids at high P-T. Despite the oxidized nature of the investigated carbonate-silicate melt (i.e., among the candidate metasomatic liquids at mantle conditions), the  $f_{\text{O}_2}$  calculated at carbon saturation (Huijzen, 2005) at the experimental P-T conditions (Table 1) vary between  $-3.2$  and  $-4.6$  as normalized to the fayalite-magnetite-quartz reference buffer (Ballhaus et al., 1991). These  $f_{\text{O}_2}$  values indicate the lowest bound at which the dominant  $\text{CO}_2$  would coexist with CO and  $\text{H}_2\text{O}$  volatile species in equilibrium with graphite based on our spectroscopic observations. The low  $\text{Fe}^{3+}/\sum\text{Fe}$  measured in one of the glasses can be explained in light

of the formation of reduced  $\text{M}_x(\text{CO})_n$  carbonyl groups ( $\text{M} = \text{Fe}, \text{Pt}$ ). According to Rengan (2023) a possible redox equilibrium involving Fe can be as follows,



or,



Both reactions require the presence of CO in the carbonate-silicate liquid, the solubility of which is strongly controlled by the  $f_{\text{O}_2}$ . In particular, the CO/ $\text{CO}_2$  ratio increases as the  $f_{\text{O}_2}$  decreases until C saturation according to the equilibrium,



These conditions are likely to have been reached when the Earth's upper mantle was more reduced than today such as during the Archaean when carbonate-silicate magmas formed by redox melting of a graphite saturated peridotite rock source (Aulbach and Stagno, 2016; Stagno and Aulbach, 2021) at higher T regime. Our results highlight the potential role of carbonate-silicate melts in carrying reduced C-O-H volatiles, some of which may be responsible for mobilizing platinum group elements.

## 5. Conclusions

In this study we investigated the structure of synthetic carbonate-silicate (transitional) liquids by *in situ* multi-angle energy dispersive X-ray diffraction at pressures from 1.4 to 4.4 GPa and temperatures between 1815  $^{\circ}\text{C}$  and 1990  $^{\circ}\text{C}$ . EDXD data indicate that the intermediate range ordering decreases with increasing P and T. The structural characteristics observed at HP-T imply that viscosity of carbonate-silicate melts depends on the medium-range structural order. Then, an ambient pressure characterization of the glassy products quenched from the carbonate-silicate melt at HP-T conditions was performed by micro-Raman and reflectance- and transmittance-micro-FTIR spectroscopies. Glasses contain  $\text{SiO}_2$  amounts varying from 18.75 to 30.91 wt% and calculated volatile contents ranging between 27.72 and 6.68 wt% with  $\text{SiO}_2/\text{Al}_2\text{O}_3$  of 7.2–8.3 and  $\text{MgO}/\text{CaO}$  of 1.9–4.2. Raman spectra suggest that the silicate network in the quenched glasses is dominated by  $Q_0$  and  $Q_2$  species. A decrease in wavenumber of C-O  $\nu_3$  vibrations for  $\text{CO}_3^{2-}$  along with a significant splitting of the carbonate doublet is observed in reflectance FTIR spectra compared to what has been previously observed for the silicate and carbonate glasses, that we interpret as due to a highly distorted environment surrounding the carbonate anion. Neither evidence of  $\text{CO}_3^{2-}$  acting as a network former nor C dissolved as molecular  $\text{CO}_2$  is observed in our quenched carbonate-silicate glasses. However, we found evidence of C occurring as both free molecular CO and CO bound to Fe, Pt or FePt alloy to form metal carbonyl complexes. In *in situ* cold compression at room T of one of the quenched glasses by DAC combined with micro-reflectance-FTIR showed an increase of the relative intensity of peaks characterizing the silicate region, suggesting a more polymerized glass structure as pressure increases. Such P-driven polymerization does not occur at simultaneous high P-T conditions where the high T acts more dominantly to stretch out the tetrahedra. The reduction of the splitting of the carbonate doublet upon cold-compression suggests that T is required to cause a distorted carbonate environment.

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## CRediT authorship contribution statement

**V. Stopponi:** Writing – review & editing, Writing – original draft, Visualization, Investigation, Formal analysis. **A. D'Arco:** Investigation.

**Y. Kono:** Resources, Investigation. **F. Piccirilli:** Investigation. **B.T. Poe:** Writing – review & editing. **S. Lupi:** Supervision, Resources. **M. Nazzari:** Investigation. **L. Pappalardo:** Investigation. **G. Marras:** Investigation. **M. Zaccagna:** Resources. **C.E. Manning:** Writing – review & editing, Funding acquisition. **C. Romano:** Writing – review & editing, Resources. **V. Stagno:** Writing – review & editing, Supervision, Investigation, Funding acquisition, Conceptualization.

## Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Data availability

Data are available at <https://data.mendeley.com/datasets/xjb67vhgmr/2>.

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