

1 Revision 2 Thorite: An Oddity in Phase Stability Amongst the Zircon-Structured 2 Orthosilicates at High Pressures

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Abstract:

Synthetic thorite and huttonite, two polymorphs of ThSiO_4 , were investigated by a combination of *in situ* high-pressure synchrotron X-ray powder diffraction and *in situ* high-pressure Raman spectroscopy. The average onset pressure of the thorite-to-huttonite transition was determined to be 6.6 ± 0.2 GPa, using both techniques. The bulk moduli of thorite and huttonite were determined to be 139(9) and 246(11) GPa, respectively, by fitting their unit-cell volume data to a second order Birch-Murnaghan equation of state (EOS). Based on its bulk modulus, thorite is the most compressible zircon-structured orthosilicate, as it has the largest unit cell volume among tetravalent metal orthosilicates. The pressure derivatives of the vibrational modes of thorite were found to be consistent with those previously reported for other orthosilicates (*e.g.*, zircon, hafnon, stetindite, and coffinite), while having the smallest Grüneisen parameter. A new P-T phase diagram for ThSiO_4 is proposed, where the boundary of the thorite \rightarrow huttonite transition is: $P(T) = (7.8 \pm 0.9 \text{ GPa}) - (0.006 \pm 0.002 \text{ GPa/K})T$. Based on the new P-T phase diagram, we further estimated the enthalpy of formation of huttonite, $\Delta H_{\text{f,ox}}$, to be 0.6 ± 6.0 kJ/mol, suggesting its metastability and rare locality in nature.

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Introduction:

52 Zircon (ZrSiO_4 , $I4_1/\text{amd}$) and monazite (CePO_4 , $P2_1/n$) are important as their structures
 53 host high-field strength elements, such as Zr, Hf, rare earth element (REE), Th and U (Winter
 54 2013; Strzelecki et al. 2024). Minerals with these structures have low chemical leaching rates, high
 55 resistance to metamictization, and high thermal stability; hence, they are often used for radiometric
 56 dating based on U-Th-Pb isotopic systematics (White 2015). Interestingly, the two polymorphs of
 57 ThSiO_4 , thorite and huttonite, are isostructural with zircon and monazite, respectively. Thorite
 58 forms a complete solid solution with the isostructural uranium orthosilicate mineral, coffinite
 59 (USiO_4), which includes uranothorite phases ($\text{Th}_{1-x}\text{U}_x\text{SiO}_4$) (Guo et al. 2016; Marcial et al. 2021).
 60 In contrast, huttonite, the high temperature polymorph, possesses a monazite-type structure. Even
 61 though thorite and uranothorite are common accessory minerals, huttonite is much rarer (Pabst and
 62 Hutton 1951; Frondel 1958). The structural relation between thorite and huttonite are discussed in
 63 detail by Taylor and Ewing (1978).

The abundance of thorite and uranothorite in high P-T environments and the rarity of buttonite has always been a puzzle. While other tetravalent metal orthosilicates, such as zircon, hafnon, coffinite, and stetindite are known to exhibit a pressure-induced phase transition to the reidite structure ($I4_1/a$) (Reid and Ringwood 1969; Hazen and Finger 1979; Knittle and Williams 1993; Farnan et al. 2003; Gucsik et al. 2004; Ono et al. 2004a, 2004b; Tange and Takahashi 2004; Manoun et al. 2006; Luo and Ahuja 2008; Bose et al. 2009; Zhang et al. 2009a; Bauer et al. 2014; Stangarone et al. 2019; Strzelecki et al. 2023; Zhao et al. 2024), there has been no similar transition reported for thorite. The phase stability of thorite has only been investigated experimentally under elevated temperature and ambient pressure conditions (Taylor and Ewing 1978; Mazeina et al. 2005; Navrotsky et al. 2013). At high temperature, thorite undergoes a phase transformation to buttonite at 1483 K (Finch et al. 1964; Seydoux and Montel 1997; Mazeina et al. 2005). This likely indicates that pressure is stabilizing thorite and inhibiting its conversion to buttonite. The only insight of ThSiO_4 under elevated pressure conditions comes from a force-field computation study (Bose et al. 2009) and a more recent comprehensive density functional theory (DFT) study (Mondal et al. 2020). The results of the force-field computational study suggest that the thorite → buttonite transition is at around 1 GPa, followed by a buttonite → reidite transition at approximately 3 GPa (Bose et al. 2009). The more recent DFT study predicted that thorite only

81 undergoes the conventional thorite → reidite phase transition at 8.52 GPa, with no mention of the
82 huttonite transition (Mondal et al. 2020).

83 Therefore, the objective of this study was to experimentally determine the structures and
84 phase stabilities of thorite and huttonite under high pressure conditions through *in situ* high-
85 pressure synchrotron X-ray diffraction (XRD) and *in situ* high-pressure Raman spectroscopy. We
86 found through *in situ* high-pressure XRD that the thorite polymorph is not stable above 12.5 ± 0.5
87 GPa, in excellent agreement with 12.7 ± 0.6 GPa determined by *in situ* high-pressure Raman
88 spectroscopy. Using these determinations, a new P-T phase diagram for ThSiO₄ is proposed where
89 the boundary at the thorite-to-huttonite transition is described as: $P(T) = (7.8 \pm 0.9 \text{ GPa}) - (0.006$
90 $\pm 0.002 \text{ GPa/K})T$, suggesting that the onset pressure of the transition at 300 K is $6.0 \pm 1.1 \text{ GPa}$.

91

92 **Experimental Methods:**

93 *Sample synthesis and characterization:*

94 Two separate batches of thorite were synthesized using the hydrothermal method. The first
95 batch was studied by high-pressure powder synchrotron XRD and Raman spectroscopy, where a
96 4:1 (by volume) methanol to ethanol mixture was used as the pressure transmitting medium (PTM).
97 The sample was prepared according to the protocol of Mesbah *et al.* (2015) through modification
98 of a previous procedure (Hoekstra and Fuchs 1956; Fuchs and Hoekstra 1959). The reagents were
99 Na₂SiO₃, NaOH, and NaHCO₃. The thorium source was prepared by dissolving thorium nitrate
100 pentahydrate in 6 mol/L HCl and performing several cycles of evaporation and dissolution in 4
101 mol/L HCl to remove all traces of nitrate ions in the solution (Dacheux *et al.* 1996). The final
102 thorium concentration solution was determined by inductively coupled plasma atomic emission
103 spectroscopy (ICP-AES). The synthesis was performed under air. It consisted of first dissolving
104 Na₂SiO₃ (0.81 g, 6.66 mmol) in water and then slowly adding the corresponding thorium amounts,
105 always keeping a silicate excess of 10 mol % to ensure the complete complexation of the actinides.
106 The pH of the mixture was adjusted to 11.3 ± 0.1 by dropwise adding 8 mol/L NaOH. The solutions
107 were then buffered to pH = 8.7 ± 0.1 through the addition of NaHCO₃. The resulting white
108 gelatinous mixtures were then transferred into a 23 mL Teflon lined Parr autoclaves and placed
109 into an oven at 250 °C for 7 days. The final products were separated by centrifugation and washed
110 twice with deionized water and once with ethanol. The solids were then dried overnight in air at
111 room temperature.

112 The sample was characterized in a previous work through non-destructive means using
113 Fourier transform infrared (FTIR) and Raman spectroscopy, and laboratory based XRD (Guo et
114 al. 2016). The FTIR and Raman spectroscopic results confirmed the successful formation of a pure
115 single phased thorite, which was further confirmed by XRD (Guo et al. 2016). The chemical
116 composition of each phase was determined by electron probe microanalysis (EPMA) (Guo et al.
117 2016). The EPMA results indicate that the thorite sample has an ideal chemical composition of
118 ThSiO_4 (Guo et al. 2016). It has been shown by Guo et al. (2015) and Strzelecki et al. (2020, 2021)
119 that the presence of water can have profound effects on the structural and thermodynamic
120 responses of zircon-type phases prepared *via* hydrothermal methods. For this reason, a small
121 fraction of the samples was further characterized by thermogravimetric analysis coupled with
122 differential scanning calorimetry (TG-DSC) to elucidate the nature of any associated water. It was
123 clear from both the TG-DSC and FTIR results that all water molecules associated with the starting
124 thorite was physically adsorbed water rather than structural water. A more detailed description of
125 the synthesis can be found in Guo et al. (2016).

126 The second batch of thorite, which was compressed using anhydrous NaCl as the PTM,
127 and the huttonite were synthesized by following a procedure reported by Estevenon et al. (2018).
128 High-pressure powder synchrotron XRD was done using anhydrous NaCl as the PTM. The
129 reagents used for synthesis were Na_2SiO_3 , NaOH , NaHCO_3 , $\text{Th}(\text{NO}_3)_4 \cdot 4\text{H}_2\text{O}$, and HNO_3 . The
130 precursor solution contained 42 mmol/L Th in the presence of 3 mol.% excess Si and 420 mmol/L
131 of CO_3^{2-} . The final pH of the solution was adjusted to 8.6 ± 0.1 by dropwise addition of 4 mol/L
132 NaOH . The solution was allowed to react at 250 °C for 24 hours before being washed 6 times with
133 18.2 MΩ-cm deionized water centrifuged at 4000 rpm for 10 mins. The resulting powder was
134 allowed to dry at 150 °C for 2 hours and then sintered at 1000 °C, under air, for 24 hours, producing
135 crystalline thorite. For huttonite synthesis, the concentration of the thorium solution reached 42
136 mmol/L. The reacting medium contained a silicate of 3 mol.% excess and CO_3^{2-} concentration
137 equal to 420 mmol/L. The final pH of the solution was adjusted to 12.0 ± 0.1 by dropwise addition
138 of 4 mol/L NaOH . The solution was allowed to react at 250 °C for 43 hours before being washed
139 5 times with 18.2 MΩ-cm deionized water centrifuged at 4000 rpm for 15 mins. The resulting
140 powder was allowed to dry at 90° C overnight. The dry powder was then hand pressed into a pellet
141 form and fired at 1300 °C in air, for 10 hours, yielding crystalline huttonite.

142 XRD analysis revealed that the thorite powdered sample obtained by the second batch had
143 a minor ThO₂ impurity, while the huttonite sample had minor impurities of ThO₂ and thorite. An
144 attempt was made to purify both samples through modification of the method originally published
145 for uranothorite (Clavier et al. 2013). The method involved placing 28.91 mg of impure thorite or
146 46.27 mg of impure huttonite in contact with 50 mL of 1 mol/L HNO₃ and allowed to react in a 50
147 mL centrifuge tube. The centrifuge tube was then placed onto a Glas-Col Rugged Rotator, in
148 vertical orientation, and allowed to constantly mix for 72 hours at 60 rpm. This would allow for
149 more effective dissolution of ThO₂ by constantly exposing fresh unreacted surfaces. The solids and
150 the supernatants were then separated by decantation and the remaining solids were washed 4 times
151 with 18.2 MΩ-cm deionized water (50 mL for each stage) and once with ethanol (25 mL), allowing
152 centrifugation for 10 mins at 4000 rpm each time. The sample was then left at room temperature
153 for 48 hours to allow for all the ethanol to evaporate off. The remaining powders of thorite and
154 huttonite were hand pressed into pellets. The thorite pellet was calcined at 900 °C for 2.5 hours,
155 while the huttonite pellet was fired at 1300 °C for 8 hours. The calcined pellets were then hand
156 ground with an agate mortar and pestle. The resulting powder was analyzed by powder XRD.
157 Rietveld analysis of XRD data showed that the amount of ThO₂ impurity in the thorite and
158 huttonite did not decrease appreciably after calcination (Hubert et al. 2008), with the huttonite
159 having lesser residual thorite phase.

160

161 *In situ high-pressure synchrotron XRD:*

162 Several sets of high-pressure powder synchrotron XRD experiments were conducted at the
163 GeoSoilEnviroCARS (GSECARS) beamline 13-ID-D of the Advanced Photon Source (APS) at
164 Argonne National Laboratory (ANL). The high pressures generated during the experiments were
165 attained through application of a Princeton type symmetric diamond anvil cells (DAC). The DAC
166 were always completely sealed with Kapton tape to create secondary containment for safe handling
167 of radioactive samples. Ruby fluorescence was used to determine the pressure in the cell. To avoid
168 the damage of Kapton from absorbing the online Ruby laser, a small Mylar window was used to
169 replace the Kapton in the laser light path. The X-rays had a wavelength of 0.3344 Å (37 keV), with
170 a beam size of ~2.5 × 2.5 μm. The beams geometric parameters were calibrated using LaB₆. The
171 detector used was a Pilatus 1M cadmium telluride (CdTe) detector. The distance from sample to
172 detector was fixed at 237.1 mm.

173 The first batch of thorite, which used a 4:1 (by volume) methanol : ethanol mixture as the
174 PTM was loaded into a 150 μm laser-drilled hole at the center of a pre-indented, 40 μm thick steel
175 gasket. The culet size of the diamonds was 300 μm . A ruby sphere was also loaded into the DAC
176 to measure the pressure through the shift in its R_1 fluorescence line (Mao et al. 1986). The second
177 batch of thorite or huttonite and a piece of Pt were loaded in between two NaCl plates. NaCl was
178 utilized as the PTM and internal pressure marker. The pressure-volume (P - V) equation of state
179 (EOS) of NaCl-B1 (*Fm-3m*) (Dewaele et al. 2012), was used to determine the pressures in the cell.
180 This sample/standard/PTM assembly was loaded into an electric discharge machine (EDM) drilled
181 hole (150 μm diameter) at the center of a pre-indented (39-47 μm thickness) Re gasket. A sapphire
182 window (Edmund Optics), with a diameter of 6.35 ± 0.05 mm and a thickness of 0.50 ± 0.05 mm,
183 was placed at the opening of each DAC in case of failure of the diamonds.

184 All collected two-dimensional images were calibrated, masked, and integrated through the
185 use of Dioptas (Prescher and Prakapenka 2015). The obtained XRD patterns of thorite and
186 huttonite were analyzed by the Rietveld method using the General Structure Analysis System
187 software version II (GSAS-II) (Toby and Von Dreele 2013). The background was modelled using
188 the Chebyshev function. The starting models for the refinements for both thorite and huttonite
189 were the structures reported by Taylor and Ewing (1978). The Rietveld refinement procedures
190 have been described previously (Zhang et al. 2002; Xu et al. 2017; Guo et al. 2019; Baker et al.
191 2020; Lü et al. 2020; Strzelecki et al. 2022b, 2022a, 2023) .

192

193 *In situ* high-pressure Raman spectroscopy:

194 Raman spectroscopic measurements were performed on the first batch of thorite using a
195 Horiba HR Evolution Labram Raman system equipped with a 532 nm laser, and an Olympus 20 \times
196 long-working distance objective lens was used to visualize the sample and focus the laser into the
197 sample chamber. All spectra were collected in the 100-1400 cm^{-1} range. The maximum laser output
198 was 100 mW but was attenuated before interacting with the sample. The system was equipped
199 with an 1800 gr/mm grating which yielded an effective 0.3 cm^{-1} resolution. The collected spectra
200 were corrected by subtracting the background and fitted using a Lorentz-type function. For *in situ*
201 high-pressure Raman spectroscopy, a Princeton-style symmetric diamond anvil cell (DAC) with
202 Type IIb, ultra-low fluorescence 300 μm culet diamonds (Almax) were used. The sample was
203 loaded in a drilled hole (125 μm diameter) at the center of a pre-indented (50 μm thickness)

204 stainless-steel gasket. Ruby spheres were placed into the sample chamber for continuous pressure
205 determination, before and after collecting a Raman spectrum, by the pressure shift of the R1
206 fluorescence line (Mao et al. 1986). The PTM used was a 4:1 methanol : ethanol mixture. Raman
207 spectra were collected on compressing the sample from ambient to 26.4 ± 1.3 GPa. Above $15.5 \pm$
208 0.8 GPa, both the deviatoric stresses in the DAC, caused by the 4:1 ethanol methanol mixture, and
209 the sluggish kinetics of the thorite-to-huttonite transition yielded broad Raman spectra that could
210 not be deconvoluted using Gaussian functions.

211

212 **Results:**

213 *XRD of thorite using PTM of 4:1 methanol : ethanol mixture:*

214 Figure S1 shows a plot of the collected XRD patterns as a function of pressure. The onset
215 of the thorite \rightarrow huttonite transition was 5.1 ± 0.3 GPa, which is evident by the disappearance of
216 the (220) peak associated with thorite. The phase transition was deemed complete at 13.2 ± 0.7
217 GPa when no more peak could be attributed to thorite (Figure S1). Rietveld analysis was performed
218 on all of the patterns collected from 1.5 to 10.3 GPa, which yielded R_{wp} values ranging from 4.74%
219 to 6.65% (Table S1). No new peaks assigned to the huttonite phase were detected until 11.2 ± 0.6
220 GPa, and thus only the thorite phase was included in the Rietveld analysis. Above 10.3 ± 0.5 GPa,
221 the resulting patterns were too broad and distorted to reliably fit the huttonite structure (Figure S1).
222 This is probably due to the loss of hydrostaticity of the 4:1 methanol : ethanol mixture at high
223 pressure. Thus, no unit cell parameters for huttonite were derived from this set of XRD data.

224

225 *XRD of thorite using PTM of NaCl:*

226 Figure 1 plots the XRD patterns as a function of pressure. The resulting refinements for
227 thorite (with NaCl and ThO₂ as additional phases), using NaCl as the PTM, yielded R_{wp} values
228 ranging from 2.2% to 3.2% (Table S2). The small spot size ($\sim 6.3 \mu\text{m}^2$) of the X-ray beam allowed
229 for multiple positions in the sample chamber to be measured at each pressure step. However, spots
230 with both Pt and ThSiO₄ were not found so the pressure in the cell was determined only using the
231 P-V EOS of NaCl. The refined unit cell parameters of thorite are shown in Figure 2. With NaCl as
232 the PTM, thorite was found to be stable up to 6.5 ± 0.3 GPa (Table S2). The onset of the thorite
233 \rightarrow huttonite transition was 6.9 ± 0.4 GPa. This is apparent by the appearance of a new peak,
234 associated with the (011) diffraction maxima of huttonite, which appears at $2\theta = 4.608^\circ$. This peak

235 slowly grew in intensity following the compression up 9.0 ± 0.5 GPa. The phase transition was
236 deemed complete at 12.5 ± 0.6 GPa, as reflected by the absence of visible diffraction maxima of
237 thorite at that pressure (Figure S4). Due to the low intensity of the (011) peak of huttonite, only
238 NaCl and ThO₂ were included as additional phases in the Rietveld analysis along with thorite,
239 from 6.9 to 9.0 GPa.

240

241 *XRD of huttonite using PTM of NaCl:*

242 The resulting refinements for huttonite, using NaCl as the PTM, yielded R_{wp} values ranging
243 from 3.8% to 6.0% (Table S3). The refined unit cell parameters of huttonite are listed in Table S3
244 and shown in Figure 2. No additional phase transitions were observed during compression. With
245 increasing pressure, the unit cell of huttonite contracts (Figures 2 and 3). Furthermore, unlike
246 thorite, there were no emergence or disappearance of XRD peaks in the pressure range
247 investigated, suggesting no phase transition.

248

249 *Raman spectroscopy of ThSiO₄ using PTM of 4:1 methanol : ethanol mixture:*

250 The Raman spectra of ThSiO₄ collected during the compression to 26.4 ± 1.3 GPa are
251 shown in Figure 4. The vibrational peak positions for thorite under ambient pressure are listed in
252 Table 1 alongside other zircon structure-type orthosilicates (Dawson et al. 1971; Syme et al. 1977;
253 Hoskin and Rodgers 1996; Geisler et al. 2006; Clavier et al. 2014; Estevenon et al. 2018, 2019a,
254 2019b, 2020; Strzelecki et al. 2021, 2023). The onset pressure of the thorite \rightarrow huttonite transition
255 was 7.7 ± 0.4 GPa, as indicated by a shoulder emerging around 870 cm^{-1} . This shoulder is likely
256 attributed to the v₃ antisymmetric stretching mode of the huttonite structure (Jin and Soderholm
257 2015). Thorite was found to be stable up to 12.7 ± 0.6 GPa, in good agreement with results from
258 XRD analysis. Above 15.5 ± 0.8 GPa, the vibrational bands became very broad, where the
259 broadness was likely due to the deviatoric stress as the result of the 4:1 methanol:ethanol PTM
260 reaching its hydrostatic limit (Klotz et al. 2009) and also partially due to lesser crystallinity of the
261 pressure-induced phase (Bauer et al. 2014). Only the vibrational bands attributed to the thorite
262 structure could be deconvoluted and the pressure dependency of these bands were assessed (Table
263 2).

264 Theoretically, the *I4₁/amd* space group has seventeen Raman-active vibrational modes
265 belonging to the D_{4h} point group (Dawson et al. 1971; Kolesov et al. 2001; Clavier et al. 2014).

266 Among these seventeen modes, nine ($\Gamma_{\text{int}} = 2A_{1g} + 2B_{1g} + B_{2g} + 2E_g$) (Dawson et al. 1971; Kolesov
267 et al. 2001) can be assigned to the internal vibrations (or normal modes) of the SiO_4 tetrahedron,
268 and the remaining eight ($\Gamma_{\text{ext}} = 2B_{1g} + 3E_g$) (Dawson et al. 1971; Kolesov et al. 2001) assigned to
269 the external vibrations (lattice modes) (Dawson et al. 1971; Hoskin and Rodgers 1996; Kolesov et
270 al. 2001). The vibrational modes labeled as v_1 , v_2 , v_3 , and v_4 correspond to the internal vibrations,
271 where v_1 (A_{1g}) and v_3 (B_{1g}) correspond to the symmetric and antisymmetric stretching motions,
272 respectively, and v_2 (A_{1g}) and v_4 (B_{1g}) correspond to the symmetric and antisymmetric bending
273 modes, respectively (Nasdala et al. 2003). The motion of the external modes can further be
274 classified into rotational (B_g) and translational (E_g) modes (Dawson et al. 1971; Nasdala et al.
275 2003; Zhao et al. 2024). Due to the interaction of SiO_4 tetrahedra with the MO_8 dodecahedra in
276 the structure, SiO_4 tetrahedra cannot be considered strictly independent units (Syme et al. 1977).
277 As a result of these interactions, there are no reported spectra showing all seventeen Raman-active
278 modes for zircon structure-type materials (Clavier et al. 2014).

279

280 Discussion:

281 Pressure-volume equations of states:

282 Figure 2 shows the pressure dependence of the unit cell volumes of thorite and buttonite
283 from the XRD experiments and their fits to the second-order Birch-Murnaghan EOS (Birch 1978).
284 The equation describing the second-order Birch-Murnaghan EOS is:

$$285 P = \frac{3K_0}{2} \left[\left(\frac{V_0}{V} \right)^{\frac{7}{3}} - \left(\frac{V_0}{V} \right)^{\frac{5}{3}} \right] \quad (1)$$

286 in which V_0 is the zero-pressure unit cell volume, V is the cell volume at a given pressure (P), and
287 K_0 is the bulk modulus (Birch 1978). The fits were performed with the EosFit7 software (Gonzalez-
288 Platas et al. 2016) which includes uncertainties of the data points to derive a weighting scheme for
289 the fit. Uncertainties include errors from the measured pressure by ruby luminescence (Mao et al.
290 1986) and the derived unit parameters by Rietveld analyses. While other studies on zircon-
291 structured orthosilicates, such as the work of Ehlers et al. (2022), used higher-ordered Birch-
292 Murnaghan EOS to fit their experimental data, it was not used in this analysis due to the limited
293 number of data points.

294 The unit cell parameters can be further fitted to the one-dimensional form of the Birch-
295 Murnaghan EOS by replacing V with either a^3 , b^3 , or c^3 :

296
$$P = \frac{3M_{a,0}}{2} \left[\left(\frac{a_0}{a} \right)^7 - \left(\frac{a_0}{a} \right)^5 \right], P = \frac{3M_{b,0}}{2} \left[\left(\frac{b_0}{b} \right)^7 - \left(\frac{b_0}{b} \right)^5 \right], P = \frac{3M_{c,0}}{2} \left[\left(\frac{c_0}{c} \right)^7 - \left(\frac{c_0}{c} \right)^5 \right] \quad (2)$$

297 where a_0 , b_0 , and c_0 are the unit cell parameters at zero pressure and $M_{a,0}$, $M_{b,0}$, and $M_{c,0}$ are the
 298 linear moduli. The values derived for M_a , M_b , and M_c can be used to further check the previously
 299 derived K_0 in Eq. 1 and give confidence in the derived values through the following relation:

300
$$K_{0,lm} = 3 \times \left(\frac{1}{M_{a,0}} + \frac{1}{M_{b,0}} + \frac{1}{M_{c,0}} \right)^{-1} \quad (3)$$

301 Lastly, the linear compressibility (β_l) of a crystal is the relative decrease in its unit cell
 302 length l when the crystal is subjected to a unit pressure and is given by the following equation:

303
$$\beta_l = - \left(\frac{1}{l} \right) \left(\frac{\partial l}{\partial P} \right) = - \left(\frac{1}{l} \right) \left[\frac{1}{\frac{\partial P}{\partial l}} \right] \quad (4)$$

304 where l is the cell parameter a , b , or c and $(\partial P/\partial a)$, $(\partial P/\partial b)$, or $(\partial P/\partial c)$ can be derived from the
 305 above equations.

306

307 *High-pressure structure of thorite:*

308 Two P - V equations of state were determined for thorite based on the data collected from
 309 the two different PTMs (4:1 methanol : ethanol and NaCl). By using the second order Birch-
 310 Murnaghan EOS (Equation 1), the fit performed on the data collected with the 4:1 methanol :
 311 ethanol mixture PTM was based on the data collected from 1.5 ± 0.1 to 10.3 ± 0.5 GPa, and yielded
 312 a V_0 of $320.7(4)$ Å³ and a K_0 of $153(14)$ GPa (Table 3). The fit performed on the data collected
 313 with the NaCl PTM used the data collected from 3.9 ± 0.2 to 9.0 ± 0.5 GPa, and yielded a V_0 of
 314 $328.3(11)$ Å³ and a K_0 of $125(11)$ GPa (Figure 2 and Table 3). The two derived bulk moduli, using
 315 the two different PTMs, agree well with one another, giving an averaged V_0 of $324.5(6)$ Å³ and K_0
 316 of $139(9)$ GPa. These values are consistent with the values of other isostructural silicates (Table
 317 3), as there is an empirical inverse linear relationship between the V_0 and K_0 (Xu et al. 2009;
 318 Strzelecki et al. 2024). As thorite possesses the largest unit cell volume of zircon-structured
 319 orthosilicates, owing to the large ionic radius of Th⁴⁺ in an eight-fold coordination environment
 320 (Shannon 1976), it is expected that it would be the most compressible phase.

321

322 *High-pressure structure of huttonite:*

323 The elastic properties of huttonite can be determined by examining the pressure-dependent
 324 unit cell parameter data. Fitting the unit cell volume-pressure relation from 2.7 ± 0.1 to 21.1 ± 0.18

325 GPa by a second order Birch-Murnaghan EOS (Figure 2) yielded a V_0 of 296.6(3) Å³ and a K_0 of
 326 246(11) GPa. The unit cell parameters were also fitted to the one-dimensional form of the Birch-
 327 Murnaghan EOS using Equation 2. The unit-cell parameters at zero pressure for the huttonite are:
 328 $a_0 = 6.791(13)$ Å, $b_0 = 6.976(4)$ Å, and $c_0 = 6.512(11)$ Å, and the linear moduli are: $M_{a,0} = 147(16)$
 329 GPa, $M_{b,0} = 255(17)$ GPa, and $M_{c,0} = 257(42)$ GPa. The values derived for M_a , M_b , and M_c can be
 330 used to further check the previously derived K_0 and give confidence in the derived values using
 331 Equation 3. $K_{0,l}$ obtained from linear moduli for huttonite was 206(14) GPa, which compares
 332 reasonably well with the directly derived K_0 (Table 3).

333 An elastic anisotropy exists among the three crystallographic axes of huttonite (a -, b - and
 334 c -axes). The linear compressibility (β_l) of a crystal is the relative decrease in its unit-cell length l
 335 when the crystal is subjected to a unit pressure and is given by Equation 4. While all the unit cell
 336 parameters (a , b , and c) decrease with increasing pressure, their contraction rates are quite
 337 different. The rates of contraction are in the order $a < b < c$, though the rates of contraction along
 338 the b -axis and c -axis are very close. The zero-pressure compressibility of the a , b , and c axes for
 339 huttonite are $(2.26 \pm 0.25) \times 10^{-3}$ GPa⁻¹, $(1.31 \pm 0.09) \times 10^{-3}$ GPa⁻¹, and $(1.30 \pm 0.21) \times 10^{-3}$ GPa⁻
 340 ¹, respectively. Therefore, for huttonite, the unit cell along the a -axis is 1.73 and 1.74 times more
 341 compressible than along the b - and c -axes, respectively.

342

343 *Grüneisen parameters:*

344 The Raman peak positions as a function of pressure for ThSiO₄ up to 15.5 ± 0.8 GPa are
 345 shown in Figure 5. In contrast to what was found for CeSiO₄ (Strzelecki et al. 2023), the
 346 antisymmetric deformation (v_4) was discerned from the background (Figure 4) and thus included
 347 in Figure 5. The pressure derivatives of the peak positions (dv/dP) up to 15.5 ± 0.8 GPa were
 348 computed by linear fits and are listed in Table 2. All of the internal vibrational modes of the [SiO₄]
 349 tetrahedra show positive pressure dependence on the peak position (Figure 5). Overall, the
 350 individual vibrational mode shifts (dv/dP) of thorite are comparable to those of other isostructural
 351 orthosilicate minerals including zircon, hafnon, stetindite, and coffinite (Table 2) (Knittle and
 352 Williams 1993; Manoun et al. 2006; Bauer et al. 2014; Strzelecki et al. 2023).

353 Using the obtained K_0 value of thorite, in conjunction with its vibrational mode shifts, we
 354 have derived the mode Grüneisen parameters (γ_i) using the following equation:

355
$$\gamma_i = \left(\frac{K_0}{v_0} \right) \left(\frac{dv}{dP} \right) \quad (5)$$

356 The derived γ_i values for thorite are reported in Table 2. Both the value of vibrational mode shifts
357 and the bulk modulus values (Table 3) are smaller than those of other zircon-structure
358 orthosilicates and so are the γ_i . This is explained by the inverse relationship between bulk modulus
359 and unit cell volume (Xu et al. 2009). As thorite possess the largest unit cell volume of the zircon
360 structure orthosilicates, owing to the large ionic radii of Th^{4+} in an eight-fold coordination
361 environment (Shannon 1976), this is consistent with the existing empirical trends (Strzelecki et al.
362 2024).

363

364 *P-T phase diagram of ThSiO_4 :*

365 The pressure-temperature phase diagram of ThSiO_4 has been previously assessed by
366 Dachille and Roy (1964), Seydoux and Montel (1997), and Mazeina et al. (2005) using a variety
367 of experimental techniques (Figure 6). Dachille and Roy (1964) determined the P-T phase diagram
368 using a Bridgman-type opposed-anvil apparatus and studied the quenched solids. The P-T phase
369 boundary determined in Dachille and Roy (1964) can be described by $P(T) = 6.7 \text{ GPa} - (0.005$
370 $\text{GPa/K})T$, suggesting an onset pressure of 5.1 GPa for the thorite to huttonite transition (Figure 6)
371 (Dachille and Roy 1964). Seydoux and Montel (1997) used hydrothermal synthetic techniques to
372 determine a P-T boundary line described by $P(T) = 4.5 \text{ GPa} - (0.003 \text{ GPa/K})T$, which suggests a
373 much lower onset pressure of 3.6 GPa for the thorite to huttonite transition (Figure 6). Mazeina et
374 al. reassessed the P-T phase diagram and proposed a phase boundary described by $P(T) = 1.8 \text{ GPa}$
375 $- (0.001 \text{ GPa/K})T$ (Figure 6) (Mazeina et al. 2005). This reassessment was based on an improved
376 knowledge of the temperature induced phase transition of thorite-to-huttonite using differential
377 scanning calorimetry (Mazeina et al. 2005) and also the works of Dachille and Roy and Seydoux
378 and Montel (Dachille and Roy 1964; Seydoux and Montel 1997).

379 In this study, the *P-T* phase diagram was again reassessed. The reassessment was based on
380 the pressure that huttonite was first visible during our *in situ* high-pressure powder XRD and
381 Raman spectroscopy experiments. Additionally, the three data points of Dachille and Roy which
382 yielded a mixed thorite + huttonite phase were included in the reassessment (Dachille and Roy
383 1964) (Figure 6). Thus, the phase boundary of thorite-to-huttonite can be described by
384 $P(T) = (7.8 \pm 0.9 \text{ GPa}) - (0.006 \pm 0.002 \text{ GPa/K})T$, suggesting an onset pressure of the transition to
385 be $6.0 \pm 1.1 \text{ GPa}$. It can also be inferred from Figure 6 that the phase stability line determined here
386 implies a temperature of transition, at ambient pressures, of thorite-to-huttonite of 1252 K. This

387 value agrees well with the value determined by Seydoux and Montel (1997) of 1283 K even though
388 this value was not considered in our reassessment.

389 The small discrepancy, of ~30 K, in the temperature of transition of thorite-to-huttonite at
390 ambient pressures could be attributed to the sluggish kinetics of the phase transition. The reason
391 that the transition is sluggish is that the ThO_8 polyhedra in thorite become ThO_9 polyhedra in
392 huttonite (Finch and Hanchar 2003). This sluggish behavior was observed in the calorimetry work
393 by Mazeina et al., where only a fraction of the thorite was converted to huttonite even at 1773 K;
394 in fact, the authors needed to heat the thorite twice in the DSC to produce pure huttonite (Mazeina
395 et al. 2005). Furthermore, from the new phase boundary equation, we evaluate the thermochemical
396 data of thorite and huttonite with implications of their natural formation and abundance. By using
397 the Clapeyron equation $\frac{dP}{dT} = \frac{\Delta\Delta H}{T\Delta V}$, with its value from the boundary equation, and ΔV from the
398 change in volume from thorite to huttonite at pressures (volumetric data in Tables S2 and S3), we
399 can estimate the enthalpy of transition ($\Delta\Delta H$) at room temperature to be 7.0 ± 1.8 kJ/mol. With the
400 enthalpy of formation of thorite from oxides ($\Delta H_{f,ox} = -6.4 \pm 5.7$ kJ/mol) measured by high
401 temperature oxide melt calorimetry (Guo et al. 2016), $\Delta H_{f,ox}$ of huttonite can be derived to be 0.6 ± 6.0 kJ/mol.
402

403

404 **Implications:**

405 Huttonite occurs rarely in nature. The new P - T phase diagram for ThSiO_4 proposed in this
406 work offers a potential thermodynamic explanation for the rarity of natural huttonite, as the
407 stability fields of thorite and huttonite can be constrained by the updated boundary:
408 $P(T) = (7.8 \pm 0.9 \text{ GPa}) - (0.006 \pm 0.002 \text{ GPa/K})T$. This suggests that the onset pressure of the
409 thorite-to-huttonite transition is 6.0 ± 1.1 Gpa, and by using the Clapeyron equation, the enthalpy
410 of formation of huttonite is estimated to be 0.6 ± 6.0 kJ/mol at standard state conditions. This
411 close-to-zero enthalpy suggests that huttonite is only marginally stable with respect to the oxides
412 (ThO_2 and SiO_2) at room temperature but unstable relative to thorite below 6 Gpa, and then
413 becomes more stable above 6 Gpa. However, the transition from thorite to huttonite may be
414 kinetically hindered, probably due to the large energetic barrier for bonding changes from ThO_8
415 polyhedra in thorite to ThO_9 polyhedra in huttonite. This may generally explain the rarity of
416 huttonite in natural systems, and huttonite may serve as a diagnostic mineral in reconstructing P -
417 T conditions of metamorphic terrains and other geological formations.

418

419

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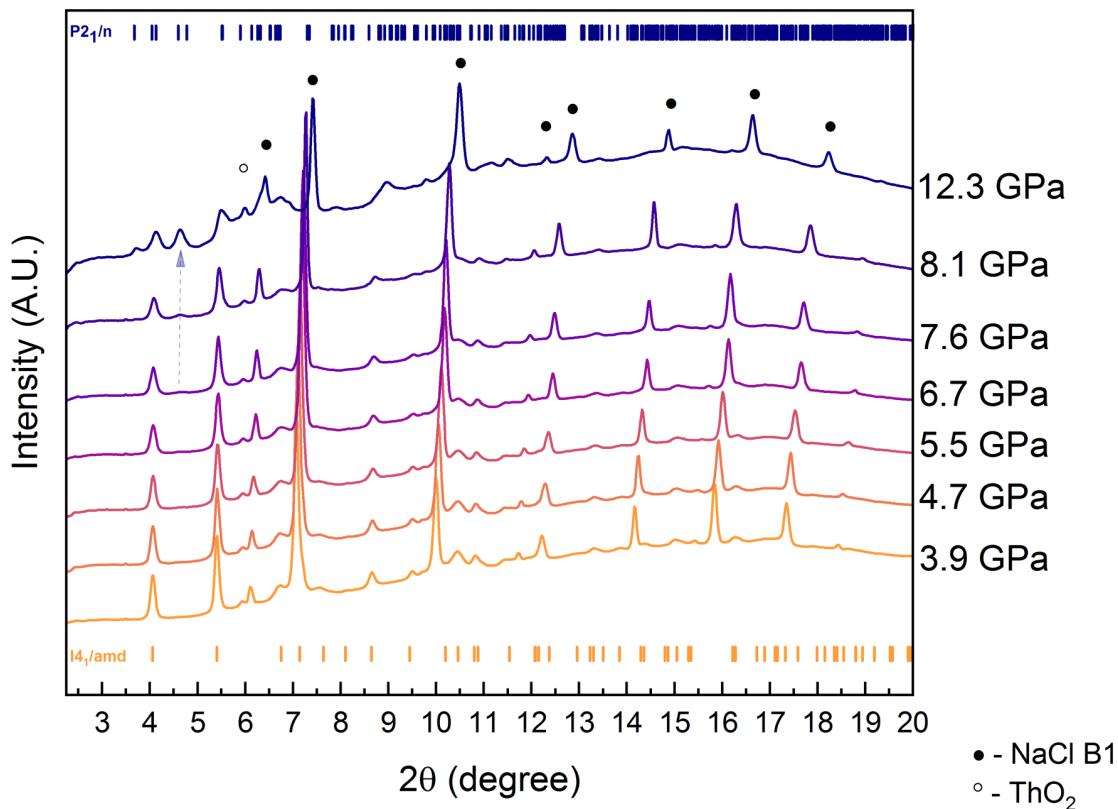
Dedication:

437 We would like to dedicate this work to Prof. Rod C. Ewing at Stanford University (also a co-author
438 of this paper), who unfortunately passed away during the revision of this paper. Prof. Ewing was
439 a world-leading mineralogist and materials scientist who pioneered the scientific field of nuclear
440 waste disposal such as using zircon-type minerals to immobilize actinides. He inspired us to pursue
441 in-depth understanding of the structures and thermodynamics of actinide orthosilicates, including
442 thorite and buttonite presented in this work, with the goal of high-fidelity prediction of their stability
443 relations and ultimate fate in a geological repository. We are deeply saddened by his passing and
444 the loss of a visionary leader in the mineralogical community.

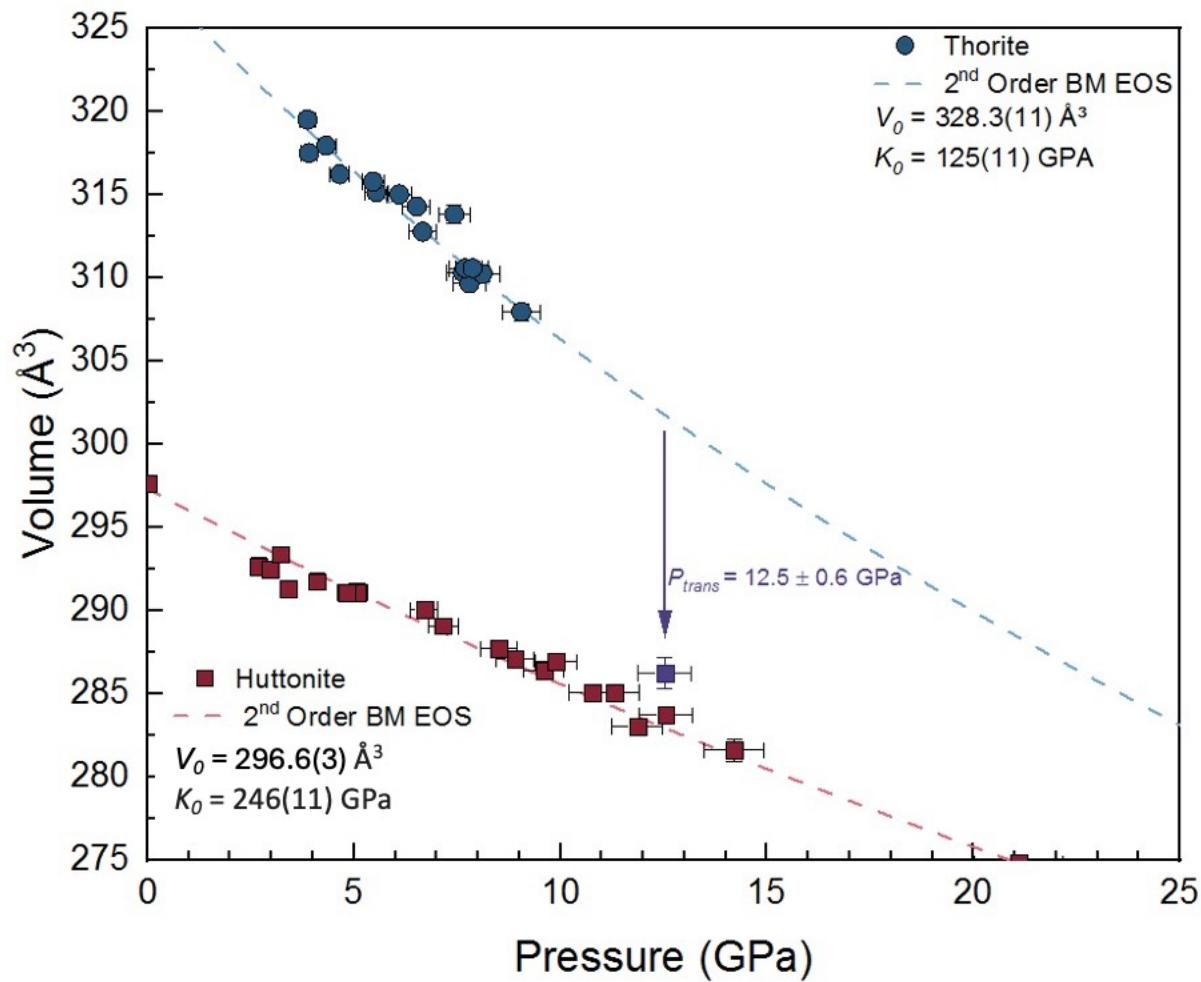
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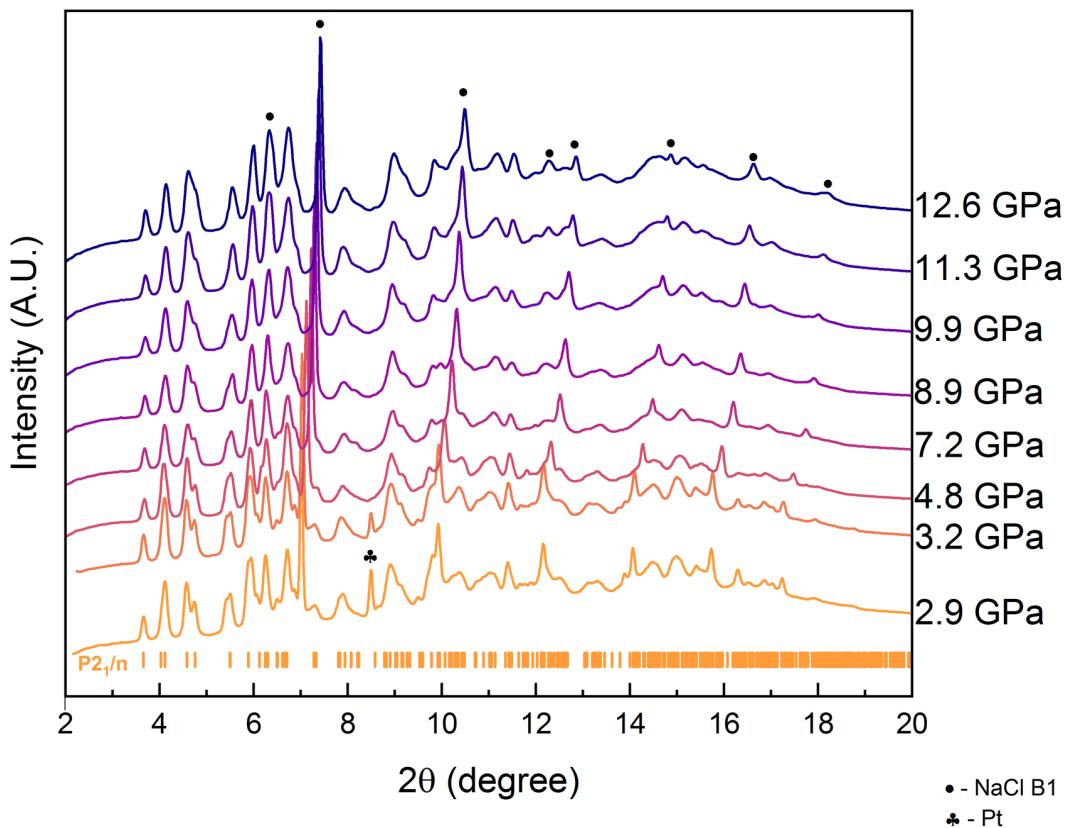
Figures:



448
449 **Figure 1:** Powder XRD patterns of thorite as a function of pressure, collected using a diamond
450 anvil cell with NaCl as the pressure transmitting medium. Ticks below and above the XRD patterns
451 indicate the positions of allowed diffraction maxima for thorite and huttonite, respectively. The
452 dashed arrow around 4.608° indicates the growth of the diffraction peak related to huttonite.
453



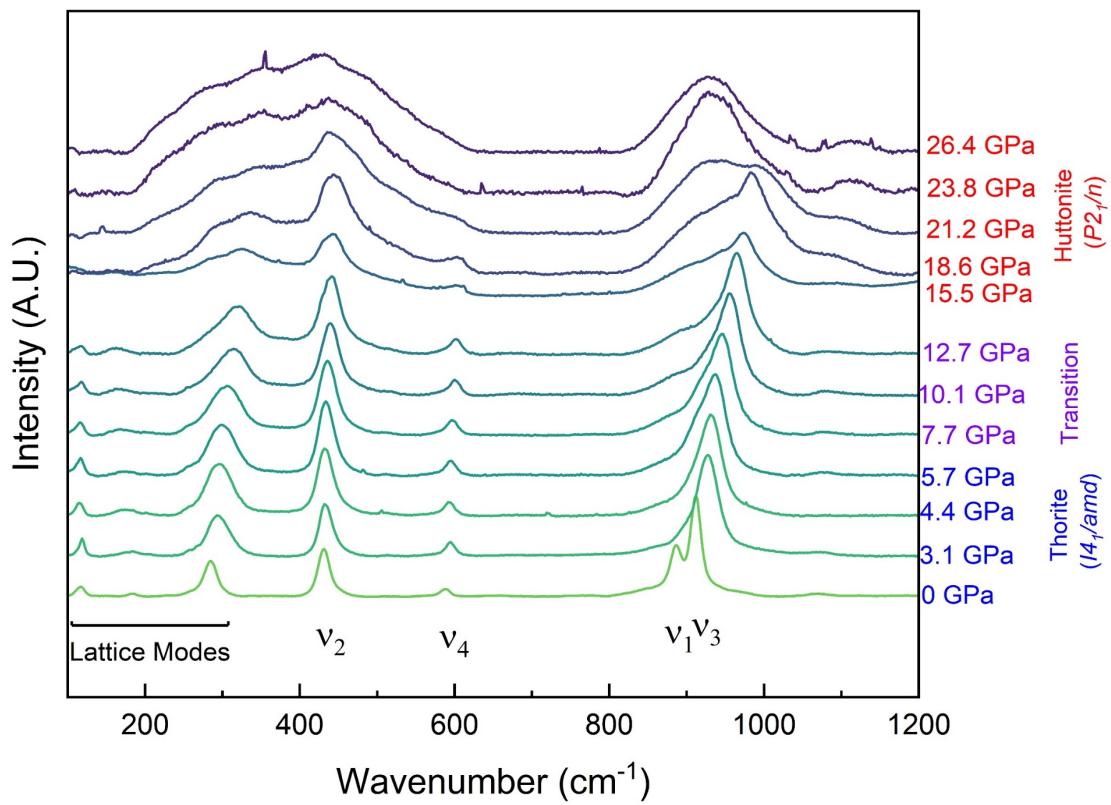
454
 455 **Figure 2:** Variation of unit-cell volumes of ThSiO_4 as a function of pressure. Blue symbols
 456 represent the low-pressure thorite structured phase ($I4_1/\text{amd}$); red symbols represent the high-
 457 pressure huttonite structured phase ($P2_1/n$); and purple symbol represents ThSiO_4 that underwent
 458 the thorite to huttonite transition. Unit-cell volumes were determined from XRD data collected
 459 using a diamond anvil cell with NaCl as the pressure transmitting medium.
 460



461

462 **Figure 3:** Powder XRD patterns of huttonite as a function of pressure, collected using a diamond
 463 anvil cell with NaCl as the pressure transmitting medium. Ticks below indicate the positions of
 464 allowed diffraction maxima for huttonite.

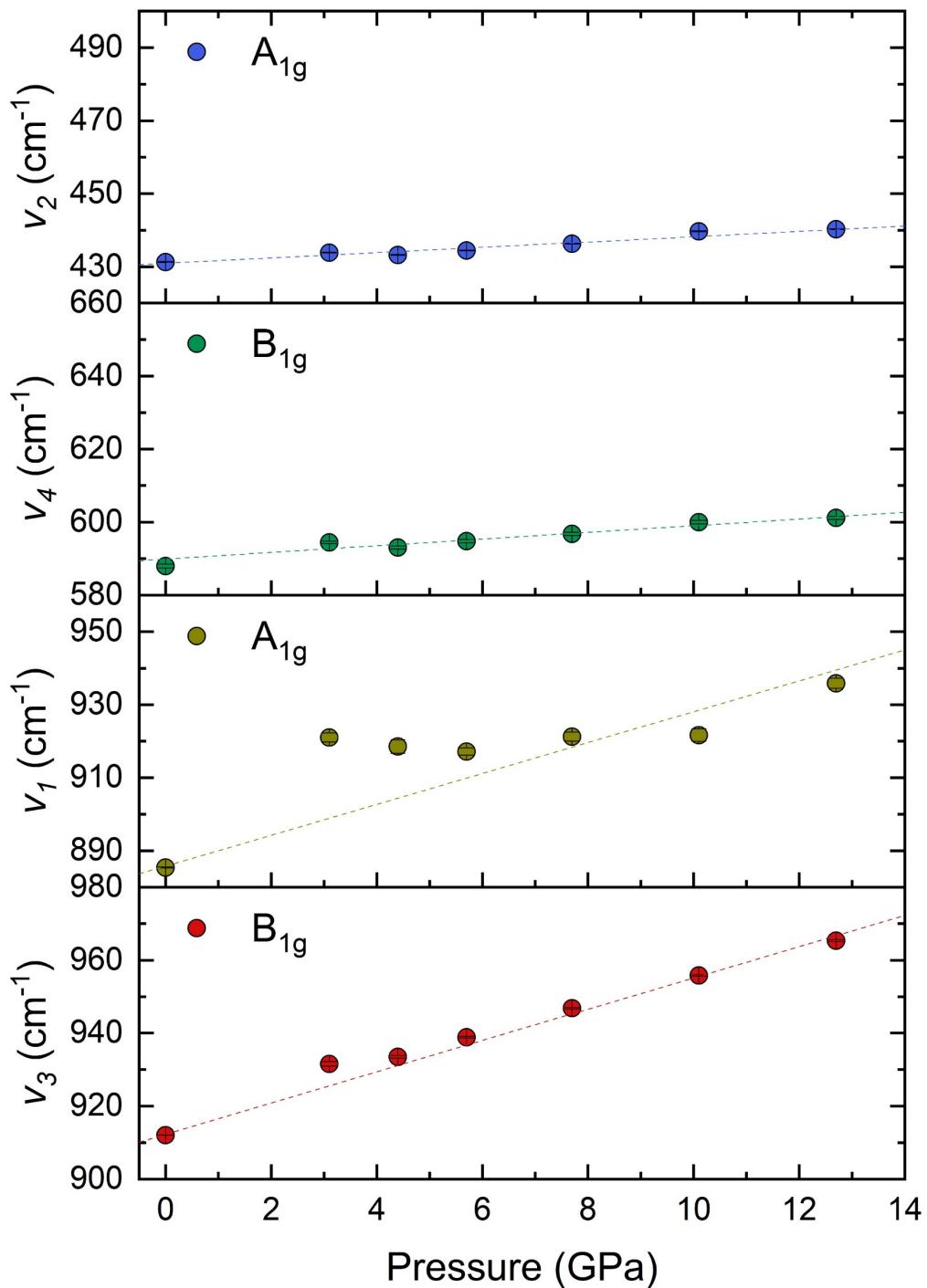
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466

467 **Figure 4:** Raman spectra of ThSiO₄ at high pressures up to 26.4 ± 1.3 GPa.

468

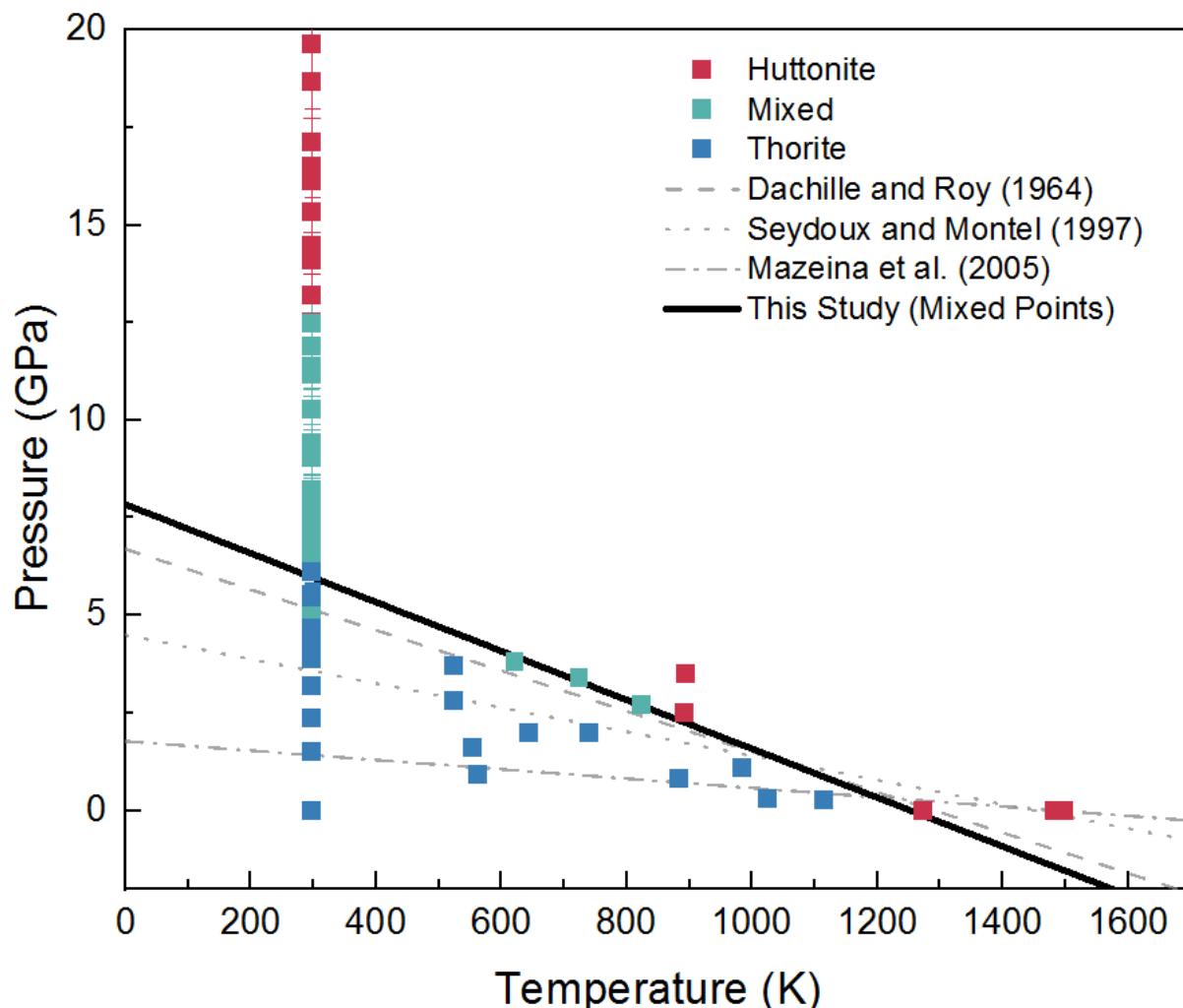


469

470 **Figure 5:** Pressure dependence of the silicate internal Raman modes of thorite up to 12.7 ± 0.6

471 GPa.

472



473

474 **Figure 6:** P-T phase diagram of ThSiO₄. Red squares represent huttonite, green squares represent
 475 a mixture of thorite and huttonite, and blue represent thorite. The lines represent the P-T phase
 476 boundary between thorite and huttonite. The data used in this figure were taken from Dachille and
 477 Roy (1964), Seydoux and Montel (1997), Mazeina et al. (2005), and this study.
 478

479 **Tables:**

480 **Table 1.** Frequencies of the internal vibrational modes of SiO_4 tetrahedra of thorite in cm^{-1} and
 481 comparison with those of some other zircon structure-type orthosilicates at ambient pressure.

	$v_1 (A_{1g})$	$v_2 (A_{1g})$	$v_3 (B_{2g})$	$v_4 (B_{2g})$	Reference
ThSiO ₄	885	431	912	588	This Work
ThSiO ₄	894	439	920	593	(Syme et al. 1977)
ThSiO ₄	887	438	914	592	(Clavier et al. 2014)
ThSiO ₄	892	439	918	596	(Estevenon et al. 2018)
USiO ₄	904	428	930	N.O.	(Geisler et al. 2006)
USiO ₄	906	424	919	591	(Clavier et al. 2014)
USiO ₄	903	424	918	592	(Strzelecki et al. 2021)
CeSiO ₄	902	417	920	594	(Strzelecki et al. 2023)
CeSiO ₄	903	431	919	592	(Estevenon et al. 2019a)
CeSiO ₄	902	431	919	592	(Estevenon et al. 2019b)
CeSiO ₄	902	416	918	593	(Strzelecki et al. 2021)
ZrSiO ₄	974	439	1008	608	(Dawson et al. 1971)
HfSiO ₄	984	448	1018	620	(Hoskin and Rodgers 1996)
HfSiO ₄	979	451	1015	627	(Estevenon et al. 2020)

482

483

484

485

486 **Table 2.** Pressure derivatives (dv/dP) of the vibrational modes and Grüneisen parameters (γ_i) of
 487 ThSiO₄ below 12.7 GPa and comparison with the values of some other isostructural
 488 orthosilicates.

489

	dv_1/dP (cm ⁻¹ /GPa)	dv_2/dP (cm ⁻¹ /GPa)	dv_3/dP (cm ⁻¹ /GPa)	dv_4/dP (cm ⁻¹ /GPa)	Method	Reference
ThSiO ₄	4.2	0.7	4.3	0.9	Experimental	This Study
USiO ₄	5.2	1.4	6.0	3.2	Experimental	(Bauer et al. 2014)
USiO ₄	5.6	1.2	5.4	1.8	DFT	(Bauer et al. 2014)
CeSiO ₄	4.9	1.2	4.8	N/A	Experimental	(Strzelecki et al. 2023)
HfSiO ₄	4.1	1.1	4.6	N/A	Experimental	(Manoun et al. 2006)
ZrSiO ₄	4.1	1.1	4.8	N/A	Experimental	(Knittle and Williams 1993)
	γ_1	γ_2	γ_3	γ_4	Method	Reference
ThSiO ₄	0.60	0.21	0.59	0.19	Experimental	
USiO ₄	1.03	0.61	1.18	0.99	Experimental	(Bauer et al. 2014)
USiO ₄	1.12	0.54	1.07	0.57	DFT	(Bauer et al. 2014)
CeSiO ₄	0.95	0.50	0.90	N/A	Experimental	(Strzelecki et al. 2023)
ZrSiO ₄	1.0	0.57	1.1	N/A	Experimental	(Knittle and Williams 1993)
USiO ₄	1.03	0.61	1.18	0.99	Experimental	(Bauer et al. 2014)
USiO ₄	1.12	0.54	1.07	0.57	DFT	(Bauer et al. 2014)

490

491 **Table 3.** Bulk moduli of thorite, other isostructural orthosilicates and huttonite.

Phase	V_0 (Å ³)	K_0 (GPa)	PTM	Reference
thorite (ThSiO ₄)	320.7(4)	153(14)	4:1 methanol:ethanol	This study
thorite (ThSiO ₄)	328.3(11)	125(11)	NaCl	This study
thorite (ThSiO ₄)	324.5(6)	139(9)	averaged	This study
coffinite (USiO ₄)	306.2(4)	188(4)	16:3:1 methanol:ethanol:water	(Zhang et al. 2009b)
coffinite (USiO ₄)	305.58(2)	181(5)	Ne	(Bauer et al. 2014)
stetindite CeSiO ₄)	300.8(3)	171(5)	Ne	(Strzelecki et al. 2023)
huttonite (ThSiO ₄)	296.6(3)	246(11)	NaCl	This Study

492

493

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