

Gate-tunable electroresistance in a sliding ferroelectric tunnel junction

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Abstract

We fabricate and measure electrically-gated tunnel junctions in which the insulating barrier is a sliding van der Waals ferroelectric made from parallel-stacked bilayer hexagonal boron nitride and the electrodes are single-layer graphene. Despite the nominally-symmetric tunnel-junction structure, these devices can exhibit substantial electroresistance upon reversing the ferroelectric polarization. The magnitude and sign of tunneling electroresistance are tunable by bias and gate voltage. We show that this behavior can be understood within a simple tunneling model that takes into account the quantum capacitance of the graphene electrodes, so that the tunneling densities of

states in the electrodes are separately modified as a function of bias and gate voltage.

Keywords: Hexagonal boron nitride, sliding ferroelectricity, graphene, tunnel junctions, two dimensional materials

Ferroelectric tunnel junctions (FTJs) are promising candidates for next-generation memory technologies due their capacity for non-volatile memory storage, non-destructive readout, and low write energy.^{1,2} In an FTJ, a ferroelectric layer which serves as an insulating tunnel barrier is sandwiched between two conducting electrodes. In order for the tunneling conductance to be sensitive to the polarization direction, thereby giving a non-zero tunneling electroresistance (TER), the junction structure must not be mirror symmetric. This has been achieved in previous work by using electrodes with different screening lengths^{3,4} or by inserting a dielectric spacer between the ferroelectric layer and one electrode.^{5,6} Making ferroelectric layers which are simultaneously thin enough to serve as tunnel barriers and stable against depolarization and wear-out processes is challenging with conventional ferroelectric materials, in part because the depolarization field from the bound charges can destabilize ferroelectricity for very thin layers.^{7,8} This has motivated interest in ferroelectrics made from van der Waals materials which show no critical thickness for ferroelectric order.^{9–15} Some van der Waals materials, e.g., transition metal dichalcogenides^{16–18} and hexagonal boron nitride (hBN)^{19–22} provide a novel form of ferroelectricity, sliding ferroelectricity, in which switching is achieved by the relative sliding motion of entire van der Waals layers. This sliding mechanism combined with the atomically pristine nature of van der Waals layers can provide higher endurance compared to non-sliding ferroelectrics.^{23,24}

Here, we demonstrate another potential advantage of assembling ferroelectric tunnel junctions from van der Waals materials – they allow the added functionality of making the TER tunable by means of electrical gating. We show that a nominally mirror-symmetric FTJ with the structure graphene electrode/bilayer hBN/graphene electrode can nevertheless still provide a substantial TER signal upon switching of the ferroelectric bilayer hBN, because the graphene electrodes can be electrically gated so that they have different densities of states for

tunneling. The resulting TER exhibits a controllable pattern with multiple sign changes as a function of gate voltage (V_G) and bias voltage across the tunnel junction (V). We explain how this pattern can be understood by taking into account the quantum capacitance of the graphene electrodes in order to track how the electron chemical potentials in the electrodes shift relative to their Dirac points as a function of V_G and V .

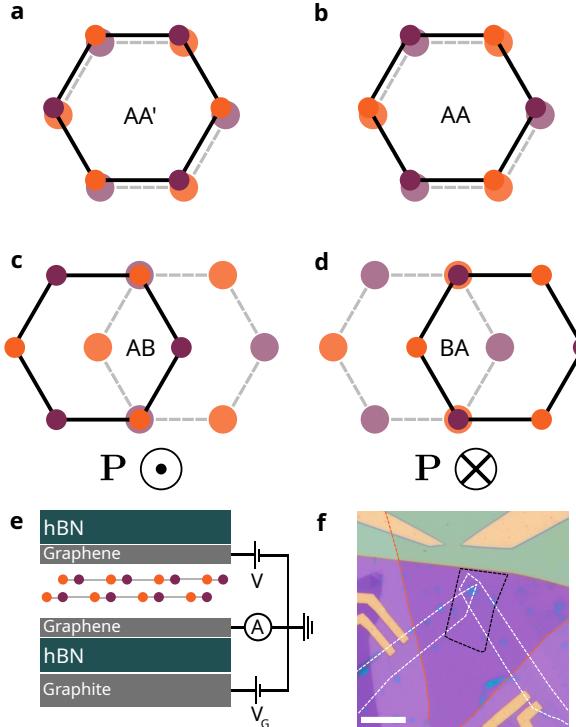


Figure 1: Sliding ferroelectric tunneling transistor. (a)-(d) Bilayer hBN allotropes. Purple (orange) circles represent boron (nitrogen) atoms while the smaller and darker (fainter and larger) circles correspond to the top (bottom) layer. (a) shows the anti-parallel AA' structure of bilayer hBN as exfoliated from bulk where the atoms of one layer fully eclipse the atoms of the layer below. (b) depicts the unstable AA parallel stacking order. The two layers are slightly offset for clarity in (a) and (b). (c)-(d) show the two ferroelectric allotropes AB and BA where the ferroelectric order points out of or into the page. (e) Diagram of device structure (not to scale). For all measurements the bottom graphene layer is grounded. (f) Micrograph of device. The dashed white, black, and orange lines outline the graphene electrodes, P-BBN, and graphite gate electrode, respectively. Scale bar is 10 μm .

For the ferroelectric tunnel barrier in our devices we use parallel-stacked bilayer boron nitride (P-BBN). When simply exfoliated from bulk hBN, native bilayer hBN (Fig. 1(a)) exhibits AA' stacking where the top layer is rotated 180° (antiparallel) relative to the bottom

layer, in which case the boron (nitrogen) atoms of the top layer fully eclipse the nitrogen (boron) atoms of the bottom layer, so that there is no net ferroelectricity. Parallel bilayer boron nitride can be obtained by tearing a monolayer hBN flake and placing one half of it on top of the other while maintaining the relative angular alignment between the two layers. Fig. 1(b) depicts P-BBN in the AA stacking configuration where the boron (nitrogen) atoms of the top layer eclipse the boron (nitrogen) atoms of the bottom layer. The AA stacking configuration is energetically unfavorable however, and the bilayer will transition to one of two degenerate ferroelectric allotropes, AB (Fig. 1(c)) or BA (Fig. 1(d)), which host opposite out-of-plane electrical polarizations. P-BBN can be switched between the AB and BA phase by applying an out of plane electric field.^{20,22} This switching is driven by the sliding motion of one entire atomic layer relative to the other by a distance of the B-N bond length. Parallel alignment is generally imperfect, in which case samples form separated domains of AB and BA stacking, and the relative sliding results in the shifting of domain walls.²⁰⁻²²

We incorporate P-BBN into an FTJ by using mechanical stacking to make heterostructures in which the P-BBN is sandwiched between two flakes of monolayer graphene which serve as the electrodes. The top of the heterostructure is encapsulated by a 75 nm thick hBN dielectric layer while the bottom is encapsulated by a 70 nm hBN dielectric layer and a 2.6 nm graphite gate electrode as shown in Fig. 1(e) and (f). The inclusion of a gate allows us to tune the electron chemical potentials of the junction electrodes. For details on the device fabrication, see Supporting Information Section S1. We measure the junction differential conductance dI/dV as a function of V at different fixed values of V_G by applying a small (1 mV) AC voltage on top of the DC bias and measuring the resulting AC current with a lock-in amplifier. All measurements are performed at $T = 4.2$ K.

We investigate the gate dependence of the TER by performing forward and backward sweeps of V at constant V_G . The left panel of Fig. 2(a) shows the differential conductance dI/dV at $V_G = -1.0$ V as V is swept forward (purple) from $V = -0.5$ V to $V = 0.5$ V and backward (orange) from $V = 0.5$ V to $V = -0.5$ V. Both forward and backwards

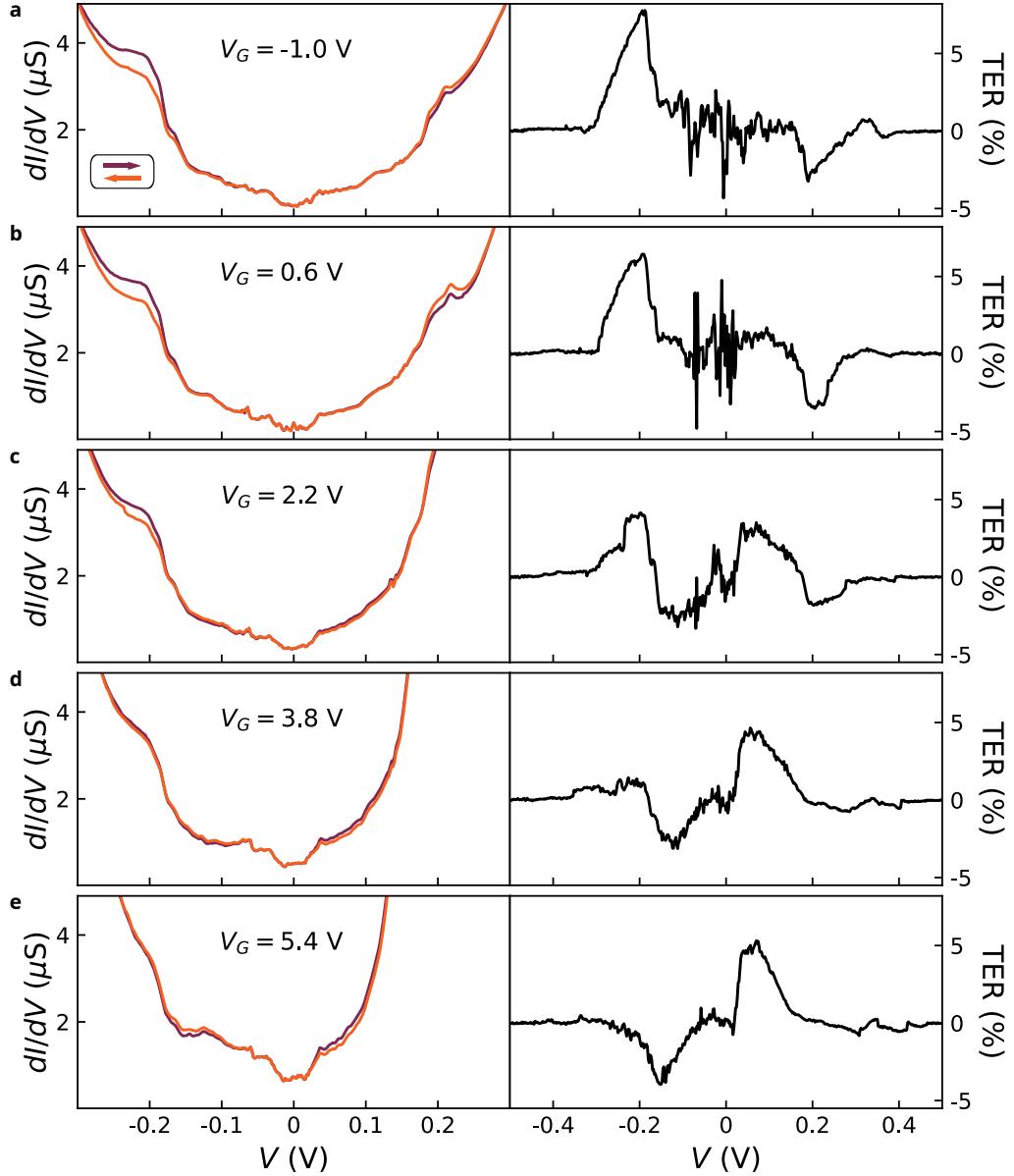


Figure 2: **Evolution of TER with gate voltage.** (a) Left panel: Tunneling conductance as a function of bias voltage at $V_G = -1.0$ V and 4.2 K for a forward (purple) and backward (orange) sweep. Right panel: TER of the left panel. (b)-(d) are the same as (a) but are measured at $V_G = 0.6$ V, 2.2 V, 3.8 V, and 5.4 V respectively.

sweeps reach a minimum of $dI/dV \sim 240$ nS when $|V| < 25$ mV and then dI/dV rises as V increases in magnitude. The increasing conductance is accompanied by step-like jumps positioned symmetrically in V . The evolution of the differential conductance with V_G is illustrated in the left-hand panels of Fig. 2(b)-(e).

The non-Ohmic behavior of dI/dV confirms that the P-BBN acts a good tunneling barrier between the conducting graphene layers. The step-like features reveal that the majority of the tunneling current is due to phonon-assisted inelastic tunneling.²⁵⁻²⁸ To highlight the importance of inelastic tunneling, in Figure 3(a) we show a color plot of the tunneling conductance (on a logarithmic scale) as a function of V and V_G on a forward bias-voltage sweep. The conductance displays prominent vertical features that correspond to step-like jumps in the conductance which persist through all gate voltages and are symmetric in V . Figure 3(b) shows the result if we average across all gate voltages to obtain G_{avg} , and Fig. 3(c) shows the derivative $|dG_{\text{avg}}/dV|$. The step-like jumps in G_{avg} near $V = \pm 28$ meV, ± 100 meV, ± 152 , and ± 184 meV lead to peaks in $|dG_{\text{avg}}/dV|$ at these voltages. Peaks near $V = \pm 28$ meV in previous studies of graphene/boron nitride/graphene tunnel junction have been attributed to an out of plane acoustic (ZA) phonon mode.^{26,28} In crystalline graphene/hBN/graphene devices like ours in which the graphene lattices are not aligned (so that the low-energy K and K' points in the two graphene layers are also mis-aligned), elastic tunneling is expected to be suppressed in the range of voltages we examine due to the need to conserve crystal momentum in elastic tunneling.^{28,29} The small non-zero conductance we observe below the first inelastic threshold is therefore likely due to disorder or impurities.

Our main focus is the hysteresis in dI/dV relative to the two sweep directions of the bias voltage, which is indicative of switching of ferroelectric order in the barrier layer. We define a TER as

$$\text{TER} = 100 \times \frac{G_f - G_b}{G_f + G_b} \quad (1)$$

where $G_{f(b)}$ is the differential conductance of the forward (backward) sweep. The right panels of Fig. 2 show the TER at different values of V_G . The sign of the measured TER depends on

both V_G and V . At $V_G = -1.0$ V, the TER reaches a maximum value of 7.8% at $V = -0.19$ V and changes sign at positive bias voltage with a negative peak of -3.2% at $V = 0.21$ V. For low biases $|V| < 0.1$ V, the TER is small relative to the peak values and noisy. The TER vanishes above approximately $V = 0.4$ V. This gives an estimate of the coercive field E_c as $E_c \approx 0.44$ V/nm. The coercive field however can vary with gate voltage due to incomplete screening of the electric field from the gate electrode,²⁰ and some domain-wall motion may occur at fields below E_c .

As V_G is tuned, the pattern of the TER signal evolves. For example, for $V_G = 2.2$ V (Fig. 2(c)), the initial TER signals observed near $V = \pm 0.2$ V decrease in magnitude and two new features emerge, a negative peak at $V = -0.11$ V and positive peak at $V = 0.07$ V. Upon further increasing the gate voltage to $V_G = 3.8$ V (Fig. 2(d)), these new peaks at $V = -0.11$ V and $V = 0.07$ V overtake the the initial two peaks from Fig. 2(a) and (b) in magnitude. The TER for $V_G = 5.4$ V (Fig. 2(e)) exhibits only a negative peak at negative bias, $V = -0.15$ V, and a positive peak at positive bias, $V = 0.07$ V – that is, signals opposite in sign compared to $V_G = -1.0$ V (Fig. 2(a)). The full dependence of the TER on V and V_G is shown as a color plot in Fig. 4(a).

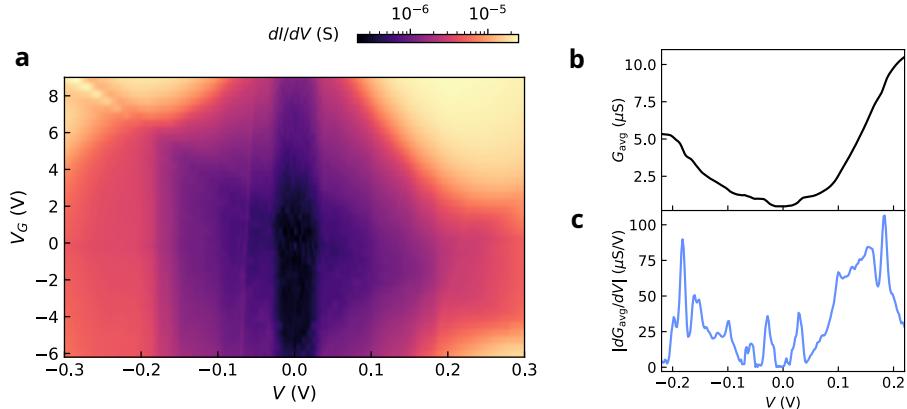


Figure 3: **Analysis of inelastic electron tunneling.** (a) Junction conductance as a function of V and V_G . V is swept from negative to positive. (b) Average conductance versus V . The average is taken over scans across all the measured gate voltages in ordered to emphasize the step-like features. (c) Derivative of average conductance. Step-like features in G_{avg} line up with peaks in $|dG_{\text{avg}}/dV|$. The positions of these peaks correspond to threshold energies for the onset of inelastic tunneling.

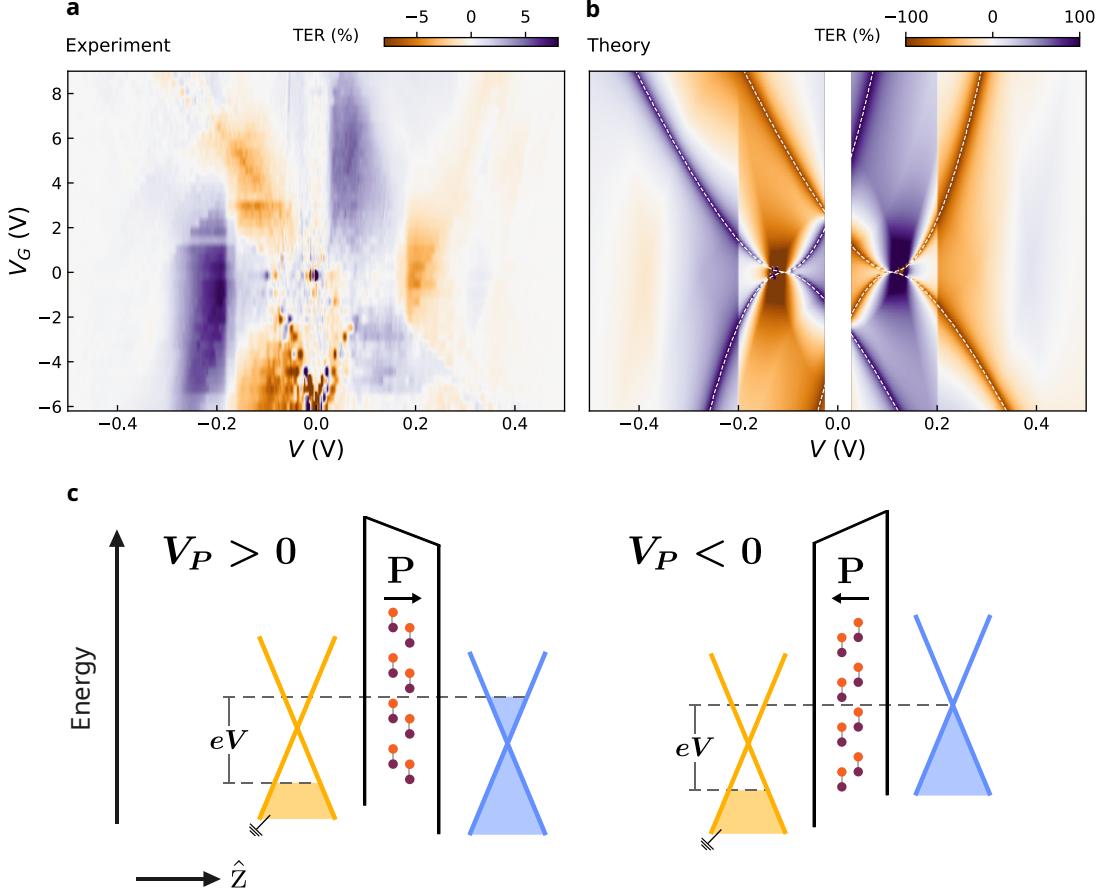


Figure 4: **Bias and gate voltage dependence of TER.** (a) Measured TER as a function of V and V_G . At each gate voltage a forward and backward sweep of V is performed to obtain the TER. (b) TER computed from the theoretical model described in the text using Eqs. (2) and (3) (see Supporting Information Sections S3 and S4 for details of the model). For color plots in (a) and (b), $\text{TER} = 0$ is set to white. The white dashed lines trace out points of charge neutrality in one of the graphene layers for the two polarization states. (c) Illustration of underlying mechanism for observed TER. The two band diagrams describe tunneling from the right (blue, top graphene) Dirac cone to the left (yellow, bottom graphene) Dirac cone across the sliding ferroelectric barrier. Both band diagrams represent the same bias and gate voltage (specifically, $V = -0.2$ V and $V_G = -2.5$ V). As the polarization switches from right, the carrier densities of the graphene layers are modified.

Although the evolution of the TER appears complicated, with multiple sign changes, this behavior can be explained qualitatively within a simple model that tracks how the electron chemical potentials in the graphene electrodes evolve as a function of changing V and V_G . Within this model, we calculate the Fermi energies of the top and bottom graphene layers, E_F^T and E_F^B (relative to the Dirac points) as a function of V and V_G by taking into

account both the geometric capacitances within the device and the quantum capacitance of the graphene layers.^{30,31} The tunneling current is then computed by considering tunneling between the two Dirac cones across the ferroelectric barrier.^{32,33} Since the tunneling signal is dominated by inelastic tunneling in our devices for the range of V in which the TER is large, in the main text we will consider a purely inelastic tunneling model. Our treatment is approximate, in that we do not require conservation of the total crystal momentum of the electron and phonon system. In the Supporting Information we consider the corresponding TER signal for elastic tunneling and achieve qualitatively very similar results – the physics behind the TER signals we observe arises from changes in the density of states for tunneling in the graphene electrodes rather than whether the tunneling is elastic or inelastic.

Within the approximations of our model, the inelastic tunneling current, I_{in} due to an excitation with threshold energy $\hbar\omega$ can be written, for positive bias voltage and $T = 0$ K, as

$$I_{\text{in}}^+ \propto e \int_{E_F^T - \frac{e\phi}{2} + \hbar\omega}^{E_F^B + \frac{e\phi}{2}} dE \rho(E - \frac{e\phi}{2})\rho(E + \frac{e\phi}{2} - \hbar\omega)\tilde{T}(E)H(eV - \hbar\omega) \quad (2)$$

and at negative bias as

$$I_{\text{in}}^- \propto e \int_{E_F^T - \frac{e\phi}{2}}^{E_F^B + \frac{e\phi}{2} + \hbar\omega} dE \rho(E - \frac{e\phi}{2} - \hbar\omega)\rho(E + \frac{e\phi}{2})\tilde{T}(E)H(-eV - \hbar\omega) \quad (3)$$

where e is the magnitude of the electron charge, ϕ is the electrostatic potential of the top graphene electrode relative to the bottom graphene electrode including the contributions from both the carriers on the graphene electrodes and the bound charge of the ferroelectric layer (note that ϕ is distinct from V because of the quantum capacitances), V_P is a polarization voltage arising from the bound charges of the ferroelectric layer, and $\rho(E) = \frac{2}{\pi\hbar^2 v_F^2} |E|$ gives the density of states in graphene where \hbar is the reduced Planck's constant and $v_F = 10^6$ m/s.³⁴ $H(eV - \hbar\omega)$ is the Heaviside step function. The WKB transmission factor can be

written in the form

$$\tilde{T}(E) = \exp \left[-\frac{2d_{\text{tun}}\sqrt{2m}}{\hbar} \sqrt{U_{\text{hBN}} - E} \right]. \quad (4)$$

In our calculations we assume $m = 0.5m_e$ for the effective mass of the hBN conduction band³⁵ (m_e is the bare electron mass), $U_{\text{hBN}} = 3$ eV,³⁶ $V_P = \pm 109$ mV,²⁰ and $d_{\text{tun}} = 0.9$ nm for the barrier thickness. For simplicity we consider the case that initially $E_F^T = E_F^B = 0$ when V , V_G , and V_P are zero. Supporting Information sections S3 and S4 describe how we solve numerically for $E_F^T(V, V_G, V_P)$, $E_F^B(V, V_G, V_P)$ and $\phi(V, V_G, V_P)$ separately for the up and down polarization states and then compute the integral for the tunneling current. The conductance is calculated as the numerical derivative of I with respect to V , and the TER is calculated as the difference in dI/dV for the up and down polarization states. Our model does not assume a particular switching voltage and instead shows the TER for the full range of V and V_G from our measurements.

For illustration purposes we compute the inelastic current assuming excitation energies at $\hbar\omega = 0.025$ and 0.2 eV using Eqs. (2) and (3) where each inelastic contribution is weighted equally. The resulting TER is shown in Fig. 4(b). Notably, the model captures all of the large sign changes for how the TER evolves with V and V_G (Fig 4(a)), indicating that gate-induced changes in the tunneling density of states in the electrodes are the primary source of asymmetry within the device that allows for a non-zero TER signal. Within the model, the largest values of TER form loci as a function of V and V_G that correspond to the electrostatic conditions for one of the graphene layers to be at charge neutrality for one of the polarization states. Switching the polarization to or from this condition leads to the largest change in the tunneling differential conductance. These loci are traced out by dashed white lines in Fig 4(b). Figure 4(c) illustrates the band diagrams of the system for the two polarization states corresponding to the large value of positive TER at $V = -0.2$ V and $V_G = -2.5$ V (the two diagrams correspond to these same values of V and V_G , with only the polarization direction switched). While our experimental data exhibit the same pattern of sign changes as in the model, the measured TER does not evolve as smoothly as a function of V and V_G as

predicted by the model. This could be because the polarization state may not remain fully fixed while sweeping V within the measurement range, or because near charge neutrality the graphene exhibits disordered regions of electron-hole puddles rather than a complete absence of carriers at the Dirac point.^{37,38}

The model predicts that the gate-induced asymmetry in the tunneling density of states is capable in principle of producing quite large values of TER, because in the model the graphene density of states is assumed to go fully to zero at the Dirac point, so that one of the differential conductances (G_f or G_b) can approach zero while the other remains large. This is unrealistic because due to the presence of disorder associated with background charges, polarization inhomogeneities, and imperfect stacking, the tunneling density of states does not go fully to zero at charge neutrality in real graphene.^{37,38} The largest TER we measure is 7.9% at $V = -1.9$ V and $V_G = -1.4$ V. This measured TER is likely also reduced from its maximum possible value because the ferroelectric polarization likely does not switch over the entire area of the tunnel junction due to the presence of pinned domains.³⁹

A reviewer asked whether the small values of TER measured near $V = 0$ might reflect that the ferroelectric polarization itself could go to zero there. Based on previous measurements on P-BBN devices (e.g., see Fig. 3 of ref. (20)) we suspect this is not the case. Rather, when our device is biased near $V = 0$ and $V_G = 0$ the symmetry of the device structure requires that the tunneling resistance must be the same for the up and down polarization states, so the TER is near zero regardless of whether or not the ferroelectric polarization has a non-zero remanence. Biasing away from $V = 0$, $V_G = 0$ is therefore required in order to obtain large TER read-outs.

In summary, our measurements demonstrate that the conductance of a ferroelectric tunnel junction with a sliding ferroelectric barrier and graphene electrodes is sensitive to the polarization state of the barrier and can generate a non-zero tunneling electroresistance (TER) when biased appropriately. The TER is tunable in sign and magnitude by varying the bias (V) and the gate voltage (V_G). The evolution of the TER with V and V_G (with its

multiple sign changes) is captured well by a simple tunneling model which takes into account how the quantum capacitance of the graphene electrodes causes the electron chemical potentials in the electrodes to shift with V and V_G . The resulting asymmetries in the tunneling density of states in the two electrodes provides the breaking of mirror symmetry required to obtain a non-zero value of TER.

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Data availability

The data used in this work are available at: [10.5281/zenodo.15159564](https://doi.org/10.5281/zenodo.15159564).

Supporting Information Available

Device fabrication, transport measurements, electrostatic model, elastic tunneling model.
Supporting figures:

- Fig. S1: Heterostructure fabrication
- Fig. S2: Electrostatics of the heterostructure
- Fig. S3: Calculation of the TER signal within an elastic tunneling model

References

- (1) Garcia, V.; Bibes, M. Ferroelectric tunnel junctions for information storage and processing. *Nature Communications* **2014**, *5*, 4289.
- (2) Park, S. H.; Lee, H. J.; Park, M. H.; Kim, J.; Jang, H. W. Ferroelectric tunnel junctions: promise, achievements and challenges. *Journal of Physics D: Applied Physics* **2024**, *57*, 253002.
- (3) Zhuravlev, M. Y.; Sabirianov, R. F.; Jaswal, S. S.; Tsymbal, E. Y. Giant Electroresistance in Ferroelectric Tunnel Junctions. *Physical Review Letters* **2005**, *94*, 246802.
- (4) Wu, J.; Chen, H.-Y.; Yang, N.; Cao, J.; Yan, X.; Liu, F.; Sun, Q.; Ling, X.; Guo, J.; Wang, H. High tunnelling electroresistance in a ferroelectric van der Waals heterojunction via giant barrier height modulation. *Nature Electronics* **2020**, *3*, 466–472.
- (5) Zhuravlev, M. Y.; Wang, Y.; Maekawa, S.; Tsymbal, E. Y. Tunneling electroresistance in ferroelectric tunnel junctions with a composite barrier. *Applied Physics Letters* **2009**, *95*, 052902.
- (6) Gao, Y. et al. Tunnel junctions based on interfacial two dimensional ferroelectrics. *Nature Communications* **2024**, *15*, 4449.
- (7) Junquera, J.; Ghosez, P. Critical thickness for ferroelectricity in perovskite ultrathin films. *Nature* **2003**, *422*, 506–509.
- (8) Kim, Y. S.; Kim, D. H.; Kim, J. D.; Chang, Y. J.; Noh, T. W.; Kong, J. H.; Char, K.; Park, Y. D.; Bu, S. D.; Yoon, J.-G.; Chung, J.-S. Critical thickness of ultrathin ferroelectric BaTiO₃ films. *Applied Physics Letters* **2005**, *86*, 102907.
- (9) Chang, K.; Liu, J.; Lin, H.; Wang, N.; Zhao, K.; Zhang, A.; Jin, F.; Zhong, Y.; Hu, X.; Duan, W.; Zhang, Q.; Fu, L.; Xue, Q.-K.; Chen, X.; Ji, S.-H. Discovery of robust in-plane ferroelectricity in atomic-thick SnTe. *Science* **2016**, *353*, 274–278.

(10) Liu, F. et al. Room-temperature ferroelectricity in CuInP₂S₆ ultrathin flakes. *Nature Communications* **2016**, *7*, 12357.

(11) Zhou, Y.; Wu, D.; Zhu, Y.; Cho, Y.; He, Q.; Yang, X.; Herrera, K.; Chu, Z.; Han, Y.; Downer, M. C.; Peng, H.; Lai, K. Out-of-Plane Piezoelectricity and Ferroelectricity in Layered α -In₂Se₃ Nanoflakes. *Nano Letters* **2017**, *17*, 5508–5513.

(12) Cui, C. et al. Intercorrelated In-Plane and Out-of-Plane Ferroelectricity in Ultrathin Two-Dimensional Layered Semiconductor In₂Se₃. *Nano Letters* **2018**, *18*, 1253–1258.

(13) Yuan, S.; Luo, X.; Chan, H. L.; Xiao, C.; Dai, Y.; Xie, M.; Hao, J. Room-temperature ferroelectricity in MoTe₂ down to the atomic monolayer limit. *Nature Communications* **2019**, *10*, 1775.

(14) Song, Q.; Occhialini, C. A.; Ergeçen, E.; Ilyas, B.; Amoroso, D.; Barone, P.; Kapeghian, J.; Watanabe, K.; Taniguchi, T.; Botana, A. S.; Picozzi, S.; Gedik, N.; Comin, R. Evidence for a single-layer van der Waals multiferroic. *Nature* **2022**, *602*, 601–605.

(15) Wang, C.; You, L.; Cobden, D.; Wang, J. Towards two-dimensional van der Waals ferroelectrics. *Nature Materials* **2023**, *22*, 542–552.

(16) Fei, Z.; Zhao, W.; Palomaki, T. A.; Sun, B.; Miller, M. K.; Zhao, Z.; Yan, J.; Xu, X.; Cobden, D. H. Ferroelectric switching of a two-dimensional metal. *Nature* **2018**, *560*, 336–339.

(17) Jindal, A.; Saha, A.; Li, Z.; Taniguchi, T.; Watanabe, K.; Hone, J. C.; Birol, T.; Fernandes, R. M.; Dean, C. R.; Pasupathy, A. N.; Rhodes, D. A. Coupled ferroelectricity and superconductivity in bilayer T_d-MoTe₂. *Nature* **2023**, *613*, 48–52, Publisher: Nature Publishing Group.

(18) Wang, X.; Yasuda, K.; Zhang, Y.; Liu, S.; Watanabe, K.; Taniguchi, T.; Hone, J.; Fu, L.; Jarillo-Herrero, P. Interfacial ferroelectricity in rhombohedral-stacked bilayer transition metal dichalcogenides. *Nature Nanotechnology* **2022**, *17*, 367–371.

(19) Li, L.; Wu, M. Binary Compound Bilayer and Multilayer with Vertical Polarizations: Two-Dimensional Ferroelectrics, Multiferroics, and Nanogenerators. *ACS Nano* **2017**, *11*, 6382–6388.

(20) Yasuda, K.; Wang, X.; Watanabe, K.; Taniguchi, T.; Jarillo-Herrero, P. Stacking-engineered ferroelectricity in bilayer boron nitride. *Science* **2021**, *372*, 1458–1462.

(21) Woods, C. R.; Ares, P.; Nevison-Andrews, H.; Holwill, M. J.; Fabregas, R.; Guinea, F.; Geim, A. K.; Novoselov, K. S.; Walet, N. R.; Fumagalli, L. Charge-polarized interfacial superlattices in marginally twisted hexagonal boron nitride. *Nature Communications* **2021**, *12*, 347.

(22) Vizner Stern, M.; Waschitz, Y.; Cao, W.; Nevo, I.; Watanabe, K.; Taniguchi, T.; Sela, E.; Urbakh, M.; Hod, O.; Ben Shalom, M. Interfacial ferroelectricity by van der Waals sliding. *Science* **2021**, *372*, 1462–1466.

(23) Bian, R.; He, R.; Pan, E.; Li, Z.; Cao, G.; Meng, P.; Chen, J.; Liu, Q.; Zhong, Z.; Li, W.; Liu, F. Developing fatigue-resistant ferroelectrics using interlayer sliding switching. *Science* **2024**, *385*, 57–62.

(24) Yasuda, K.; Zalys-Geller, E.; Wang, X.; Bennett, D.; Cheema, S. S.; Watanabe, K.; Taniguchi, T.; Kaxiras, E.; Jarillo-Herrero, P.; Ashoori, R. Ultrafast high-endurance memory based on sliding ferroelectrics. *Science* **2024**, *385*, 53–56.

(25) Amet, F.; Williams, J. R.; Garcia, A. G. F.; Yankowitz, M.; Watanabe, K.; Taniguchi, T.; Goldhaber-Gordon, D. Tunneling spectroscopy of graphene-boron-nitride heterostructures. *Physical Review B* **2012**, *85*, 073405.

(26) Jung, S.; Park, M.; Park, J.; Jeong, T.-Y.; Kim, H.-J.; Watanabe, K.; Taniguchi, T.; Ha, D. H.; Hwang, C.; Kim, Y.-S. Vibrational Properties of h-BN and h-BN-Graphene Heterostructures Probed by Inelastic Electron Tunneling Spectroscopy. *Scientific Reports* **2015**, *5*, 16642.

(27) Chandni, U.; Watanabe, K.; Taniguchi, T.; Eisenstein, J. P. Signatures of Phonon and Defect-Assisted Tunneling in Planar Metal–Hexagonal Boron Nitride–Graphene Junctions. *Nano Letters* **2016**, *16*, 7982–7987.

(28) Vdovin, E. E. et al. Phonon-Assisted Resonant Tunneling of Electrons in Graphene–Boron Nitride Transistors. *Physical Review Letters* **2016**, *116*, 186603.

(29) Perebeinos, V.; Tersoff, J.; Avouris, P. Phonon-Mediated Interlayer Conductance in Twisted Graphene Bilayers. *Physical Review Letters* **2012**, *109*, 236604.

(30) Britnell, L.; Gorbachev, R. V.; Jalil, R.; Belle, B. D.; Schedin, F.; Mishchenko, A.; Georgiou, T.; Katsnelson, M. I.; Eaves, L.; Morozov, S. V.; Peres, N. M. R.; Leist, J.; Geim, A. K.; Novoselov, K. S.; Ponomarenko, L. A. Field-Effect Tunneling Transistor Based on Vertical Graphene Heterostructures. *Science* **2012**, *335*, 947–950.

(31) Koprivica, D.; Sela, E. Resonant tunneling in graphene-ferroelectric-graphene junctions. *Physical Review B* **2022**, *106*, 144110.

(32) Simmons, J. G. Generalized Formula for the Electric Tunnel Effect between Similar Electrodes Separated by a Thin Insulating Film. *Journal of Applied Physics* **1963**, *34*, 1793–1803.

(33) Wolf, E. L. *Principles of Electron Tunneling Spectroscopy: Second Edition*; OUP Oxford, 2012.

(34) Castro Neto, A. H.; Guinea, F.; Peres, N. M. R.; Novoselov, K. S.; Geim, A. K. The electronic properties of graphene. *Reviews of Modern Physics* **2009**, *81*, 109–162.

(35) Xu, Y.-N.; Ching, W. Y. Calculation of ground-state and optical properties of boron nitrides in the hexagonal, cubic, and wurtzite structures. *Physical Review B* **1991**, *44*, 7787–7798.

(36) Watanabe, K.; Taniguchi, T.; Kanda, H. Direct-bandgap properties and evidence for ultraviolet lasing of hexagonal boron nitride single crystal. *Nature Materials* **2004**, *3*, 404–409.

(37) Martin, J.; Akerman, N.; Ulbricht, G.; Lohmann, T.; Smet, J. H.; von Klitzing, K.; Yacoby, A. Observation of electron–hole puddles in graphene using a scanning single-electron transistor. *Nature Physics* **2008**, *4*, 144–148.

(38) Xue, J.; Sanchez-Yamagishi, J.; Bulmash, D.; Jacquod, P.; Deshpande, A.; Watanabe, K.; Taniguchi, T.; Jarillo-Herrero, P.; LeRoy, B. J. Scanning tunnelling microscopy and spectroscopy of ultra-flat graphene on hexagonal boron nitride. *Nature Materials* **2011**, *10*, 282–285.

(39) Liang, J.; Yang, D.; Wu, J.; Xiao, Y.; Watanabe, K.; Taniguchi, T.; Dadap, J. I.; Ye, Z. Resolving polarization switching pathways of sliding ferroelectricity in trilayer 3R-MoS₂. *Nature Nanotechnology* **2025**, 1–7.